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# Regulating the surface of nanoceria and its applications in heterogeneous catalysis



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#### ABSTRACT

Ceria (CeO<sub>2</sub>) as a support, additive, and active component for heterogeneous catalysis has been demonstrated to have great catalytic performance, which includes excellent thermal structural stability, catalytic efficiency, and chemoselectivity. Understanding the surface properties of CeO<sub>2</sub> and the chemical reactions occurred on the corresponding interfaces is of great importance in the rational design of heterogeneous catalysts for various reactions. In general, the reversible Ce<sup>3+</sup>/Ce<sup>4+</sup> redox pair and the surface acid-base properties contribute to the superior intrinsic catalytic capability of CeO<sub>2</sub>, and hence yield enhanced catalytic phenomenon in many reactions. Particularly, nanostructured CeO<sub>2</sub> is characterized by a large number of surface-bound defects, which are primarily oxygen vacancies, as the surface active catalytic sites. Many efforts have therefore been made to control the surface defects and properties of CeO<sub>2</sub> by various synthetic strategies and post-treatments. The present review provides a comprehensive overview of recent progress in regulating the surface structure and composition of CeO<sub>2</sub> and its applications in catalysis.

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#### 1. Introduction

In recent years, due to the unique properties, ceria (CeO<sub>2</sub>) is finding prolific industrial applications [1-12]. For example, CeO<sub>2</sub>based materials have been widely explored as electrolytes in solid oxide fuel cells since their oxygen ionic conductivity is exceptionally high at the temperature range of 800–1100 K [13,14]. Fine CeO<sub>2</sub> powders are often used as polishing agents in glass and screen manufacturing [15,16] while CeO<sub>2</sub> materials are also served as an ultraviolet absorbent because of its appropriate band gap of 3.2 eV [17]. Among different utilizations, the most successful commercial application of CeO<sub>2</sub> is considered as its use in three-way catalysts (TWCs) for the elimination of toxic auto-exhaust gases, where CeO<sub>2</sub> plays an important role as an oxygen storage material in order to promote the catalytic performance of TWCs [2,18–20]. Regardless, the widespread industrial applicability of CeO<sub>2</sub> is mainly originated from its distinctive surface characteristics, which involves the reversible Ce<sup>3+</sup>/Ce<sup>4+</sup> redox pair as well as their surface acid-base properties [21-23]. Pure stoichiometric CeO<sub>2</sub> typically presents a fluorite structure with atoms arranged in a face-centered cubic unit cell with a Fm3m space group. When CeO2 is shrunk down to the nanoscale, its formation energies of structural defects would be considerably reduced, leading to the production of nonstoichiometric CeO<sub>2</sub> nanomaterials [1]. These structural defects are primarily oxygen vacancies accompanied with the generation of Ce<sup>3+</sup> species. In this case, such unique surface properties induce the intensive exploration on catalytic performances of CeO2 as the active component, promoter, and support.

Generally, heterogeneous catalysis involves (1) the adsorption of at least one of the reactants on the less coordinated surface sites of solid catalysts, (2) the activation of the adsorbed reactants, (3) the subsequent formation of the intermediate species and the adsorbed products through surface reactions catalyzed by the active sites, and (4) the final release of the adsorbed products from catalyst surfaces and the recovery of the surface active sites. Therefore, any

change in the surface properties of solid catalysts would substantially affect the entire catalytic process, which dictates the corresponding catalytic efficiency and chemoselectivity. For instance, the structural defects of CeO2-based catalysts is observed to have a significant effect on their intrinsic catalytic capabilities and activities for many desired reactions [4]. Inspired by the motivation to modulate and optimize the surface properties of CeO2, the synthesis of CeO2 nanostructures with controllable chemical compositions, uniform sizes, and well-defined morphologies has become one of the prime topics in the scientific community. It has been found that the synthetic strategies by considering the pressure, temperature, surfactants, and additives, etc. play essential roles in determining the surface properties of as-synthesized CeO2 nanostructures. Moreover, the technological applications of CeO<sub>2</sub> nanostructures have also been comprehensively discussed [4,24]. However, even though there are extensive efforts, including chemical doping and various post-treatments, have been made to regulate the structural defects of CeO2, there still remains a lack of an effective approach to realize the controllable modulation of the surface properties of CeO<sub>2</sub>.

Until now, the synthesis of CeO<sub>2</sub> nanostructures and their applications in catalysis and biology have been well summarized in the literature [1,3–7,9–11,25–33]; nevertheless, the modulations of surface physicochemical properties of CeO<sub>2</sub> have not been surveyed in details. In this review, we present a comprehensive summary on the approaches to regulate surface properties of nanoscaled CeO<sub>2</sub> and their applications in advanced catalysis. Following an introduction, a brief summary of the synthesis of CeO<sub>2</sub> nanostructures and their effects on the surface properties of as-synthesized CeO<sub>2</sub> are thoroughly discussed. It includes the conventional synthetic methods as well as the recently developed novel synthetic approaches. Regulating the surface properties of CeO<sub>2</sub> by post-treatments and chemical doping is also systematically described. More importantly, the surface chemistry of CeO<sub>2</sub> nanomaterials manipulated by various strategies for advanced catalysis, including

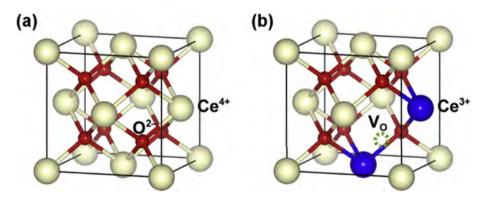


Fig. 1. (a) Ideal crystal structure of CeO<sub>2</sub>. (b) Crystal structure of CeO<sub>2</sub> in the presence of one oxygen vacancy accompanied with two generated Ce<sup>3+</sup> species.

gas-solid, liquid-solid, and biometric reactions are as well carefully reviewed. Based on all these discussion, the insightful summary and perspective outlook are then presented at the end.

#### 2. Crystal structures of CeO<sub>2</sub>

Typically, stoichiometric  $CeO_2$  adopts a cubic fluorite structure with a face-centered cubic unit cell (a = 0.5411 nm, JCPDS 34–394). In this structure, each  $Ce^{4+}$  ion has the coordination number of eight and is centered in the cubic cell of eight  $O^{2-}$  centers (Fig. 1a). Each  $O^{2-}$  ion is coordinated to four  $Ce^{4+}$  centers. Because of the reversible surface  $Ce^{3+}/Ce^{4+}$  redox pair, the nonstoichiometric  $CeO_2$  is generally formed by the release of oxygen and the subsequent formation of oxygen vacancy within the crystal structure can be understood as shown in Equation (1).

$$2Ce_{Ce}^{X} + O_{O}^{X} \rightarrow V_{O}^{**} + 2Ce_{Ce}' + \frac{1}{2}O_{2}(g)$$
 (1)

At the same time, Fig. 1b shows an illustration of the structural defects of  $CeO_2$ . With the formation of an oxygen vacancy in the lattice, the removal of an oxygen atom would leave two electrons localized in the two cerium cations and alters their oxidation states from  $Ce^{4+}$  to  $Ce^{3+}$ , accordingly. However, the precise spatial locations of the missing oxygen atom and the two cerium cations cannot be easily identified [34–36]. The recent scanning tunneling microscopy (STM) experiments and PBE + U theoretical calculations revealed that both  $Ce^{3+}$  cations or at least one  $Ce^{3+}$  ion is not adjacent to the site of oxygen vacancy [34,35]. Since the size of  $Ce^{3+}$  is larger than that of  $Ce^{4+}$ , the formation of oxygen vacancy would also induce certain lattice distortion, affecting the corresponding charge distribution on the surface or in the bulk of  $CeO_2$  [36].

Moreover, the formation of oxygen vacancy would become extremely prominent when the size of CeO2 crystals is shrunk down to the nanoscale. The presence of surface defects that are primarily oxygen vacancies can enrich the surface chemistry of nanostructured  $CeO_2$ . In addition to the reversible  $Ce^{3+}/Ce^{4+}$  redox pair, the surface of CeO<sub>2</sub> is also demonstrated with the multifunctional behavior with the coexistence of surface acidic and basic sites. The terminated OH groups are typically bases on the surface while the Ce<sup>4+</sup> ions have both Brønsted and Lewis acidic characteristics. Therefore, the multifunctionalized surface properties of CeO<sub>2</sub> nanostructures can trigger numerous practical applications in catalysis. Usually, the catalytic performance of CeO<sub>2</sub>-based catalysts hinges upon how the CeO<sub>2</sub> surface may be effectively reduced and how the oxygen be efficiently stored; in other words, the performance is highly depended on the effectiveness of the material repeatedly passing through Ce<sup>3+</sup>/Ce<sup>4+</sup> redox cycles [37]. Also, the ability of nanoceria to switch between different oxidation states is largely determined by the ratio of  $Ce^{3+}/Ce^{4+}$  and the concentration of oxygen vacancies. As a result, many efforts have been made for controlling the concentrations of surface Ce<sup>3+</sup> and oxygen vacancies by various synthetic strategies, chemical doping schemes, and post-treatment methods of CeO<sub>2</sub> [2,4,38].

In this regard, many technologies have been developed and devoted to understand the surface properties of CeO<sub>2</sub> and their corresponding catalytic characteristics considering the single crystal-based material with well-defined crystal structures as model catalysts under ultra-high vacuum (UHV) conditions [27,39–42]. These techniques include STM, X-ray photoelectron spectroscopy (XPS), Fourier transform infrared spectroscopy (FTIR), X-ray absorption spectroscopy (XAS), and calorimetry, etc, which have been already well summarized in previous reviews [30,33,43]. The catalytic interaction between well-defined CeO<sub>2</sub> surfaces and small molecules (e.g., CO, H<sub>2</sub>, H<sub>2</sub>O, CO<sub>2</sub>, NO<sub>x</sub>, and CH<sub>4</sub>) can also be revealed by these characterization approaches [30,33,43,44]. All

these studies would provide important insights into the catalytic mechanism and pave plausible approaches for the design of efficient and robust catalysts. However, practical catalytic reactions generally occur under high pressure with nanoscaled materials as catalysts. Thus, these investigations often encounter the so-called material gap and pressure gap [44]. The material gap is referred to the size effect encountered when the catalyst is scaled to nanodimension, as compared with the two-dimensional (2D) planar single crystal surface. For example, the active sites in the corners and edges of the nanoscaled catalysts cannot be simulated precisely along with the single crystal surface. Similarly, the pressure gap is denoted as the surface reconstruction and adsorption configuration of reactants on the catalyst surface under UHV and high pressure conditions. Both gaps might induce very different electronic structures of the catalytic sites, and even different catalytic active sites and various adsorption configurations of reactants under the practical and UHV conditions.

For characterizations of the surface properties of CeO<sub>2</sub> nanostructured catalysts, the surface Ce<sup>3+</sup> fraction, concentration of oxygen vacancy, and oxygen storage capacity (OSC) are commonly used as important parameters in the field of catalysis. In this case, the XPS technique is widely used to study the chemical states of heterogeneous catalysts, which can provide a quantitative analysis on the surface properties of a solid. For instance, by fitting the XPS peak of Ce 3d core level and integrating all the respective peaks, the surface Ce<sup>3+</sup> fraction of CeO<sub>2</sub> nanocrystals can be determined [45]. Also, the Raman spectrum of CeO<sub>2</sub> usually shows two characteristic bands of  $CeO_2$  at ~460 and ~600 cm<sup>-1</sup>. The strong band at ~460 cm<sup>-1</sup> can be attributed to the Raman-active vibrational mode (F2g) of the fluorite-type structure, while the band at  $\sim 600 \, \mathrm{cm}^{-1}$  is due to the existence of intrinsic oxygen vacancies [37,45]. Thus, the ratios of integral areas of the two peaks can be qualitatively compared with the concentrations of oxygen vacancies among various CeO2 catalysts. The concentration of surface oxygen vacancy (N in cm<sup>-3</sup>) can also be semi-quantitatively calculated from the grain size of CeO<sub>2</sub> nanostructures, which can be obtained by Raman spectroscopy, as shown in Equations (2)–(4) [46]:

$$d_g(nm) = \frac{51.8}{\Gamma(cm^{-1}) - 5} \tag{2}$$

$$L(nm) = \sqrt[3]{\left(\frac{\alpha}{2d_g}\right)^2 \left[\left(d_g - 2\alpha\right)^3 + 42d_g^2\alpha\right]}$$
 (3)

$$N = 3/4\pi L^3 \tag{4}$$

where  $\Gamma$  (cm<sup>-1</sup>) is the half-width at half-maximum of the strong Raman peak of CeO<sub>2</sub> at ~460 cm<sup>-1</sup>,  $d_g$  is the calculated grain size of CeO<sub>2</sub> from the Raman spectrum, L is the calculated average distance between two lattice defects, and  $\alpha$  is the radius of CeO<sub>2</sub> unit (0.34 nm).

In addition, the reversible Ce<sup>3+</sup>/Ce<sup>4+</sup> redox pair suggests the capability of CeO<sub>2</sub> to switch between releasing and storing of oxygen. This surface property is defined as the OSC, which can be used to quantitatively evaluate the capability of CeO<sub>2</sub> catalysts in releasing and storing oxygen [47,48]. The temperature-programmed reduction by reductants (*e.g.*, CO, H<sub>2</sub>, and hydrocarbon) and re-oxidation by oxidants (*e.g.*, O<sub>2</sub>, and NO) can give the OSC values of CeO<sub>2</sub> at a specific temperature [49–54]. Notably, the OSC parameter of CeO<sub>2</sub>-based materials is critical for TWCs in the automobile converters, where a high value of OSC can promote the oxidation of CO and hydrocarbon and the reduction of NO<sub>x</sub> [47,55,56].

**Table 1**Summary of synthetic methods for CeO<sub>2</sub> catalysts.

#	Method	Solvent	Reagent	Stabilizer/ template	Conditions	Production description	Ref.
1	Hydrothermal	Water	NaOH, Ce(NO <sub>3</sub> ) <sub>3</sub> ·6H <sub>2</sub> O	_	100 °C 24 h in a Pyrex bottle	Porous ceria nanorods, $8 \times 60 \text{ nm}$	[37]
2	Hydrothermal	Water	$Ce(NO_3)_3 \cdot 6H_2O$ , ammonia, hydrogen peroxide	_	160 °C 12 h in autoclave 250 °C for different times, followed by heat treatment at different temperatures	Nanoparticles, 10–15 nm	[65]
3	Hydrothermal	Water	NaOH, Ce(NO <sub>3</sub> ) <sub>3</sub> · $6H_2O$	_	100, 140, 180 °C with different alkali concentrations for 24 h	Polyhedra, $11 \pm 1.8 \text{ nm}$ Nanorods: $(9.6 \pm 1.2) \times (50-200) \text{ nm}$ Nanocubes: $36.1 \pm 7.1 \text{ nm}$	[68]
4	Hydrothermal	Water	$Ce(NO_3)_3 \cdot 6H_2O$ , urea	_	160 °C for different times	<del>-</del>	[69]
5	Hydrothermal	Water	Ce(NO <sub>3</sub> ) <sub>3</sub> ·6H <sub>2</sub> O,Na <sub>3</sub> PO <sub>4</sub> ·6H <sub>2</sub> O	-		Nanooctahedra, 100–200 nm Nanorods, hundreds of nanometers	[70]
6	Hydrothermal	Water	$Ce(NO_3)_3 \cdot 6H_2O$ , urea	СТАВ		triangular microplates: edge length of 300 nm-1 $\mu$ m, thickness of 100–200 nm	[72]
7	Hydrothermal	Water	Ce(NO <sub>3</sub> ) <sub>3</sub> ·6H <sub>2</sub> O, NaOH, CeCl <sub>3</sub> ·7H <sub>2</sub> O, NaNO <sub>3</sub> , NaBrO <sub>3</sub> , NaBr, NaI, and Na <sub>2</sub> SO <sub>4</sub>	_	150 °C for different times	Different morphology when different anions are added during the hydrothermal process, including nanorods and nanocubes	[73]
8	Simultaneous polymerization- precipitate reaction, metamorphic reconstruction, and hydrothermal process		$Ce(NO_3)_3 \cdot 6H_2O$ , glucose, acrylamide, ammonia	_	$180^{\circ}\text{C}$ for 72 h, followed by annealing at $600^{\circ}\text{C}$ for 6 h and $400^{\circ}\text{C}$ for 4 h	Flowerlike microspheres with diameter of several micrometers	[75]
9	Solvothermal	Toluene	Ce(NO <sub>3</sub> ) <sub>3</sub> ⋅6H <sub>2</sub> O, tert- butylamine	Oleic acid	180 °C for 24 h	Nanocubes, 15.65 nm	[76]
10	Solvothermal	Ethanol, water	$Ce(NO_3)_3 \cdot 6H_2O$ , TMAH	PVP	200 °C for 18 h	Nanocubes, 9—17 nm Truncated octahedra, 17 nm	[77]
11	Thermal decomposition	Oleic acid, oleylamine, 1- octadecene	Cerium acetate hydrate, sodium oleate	Oleic acid, oleylamine, 1- octadecene	320-330 °C for 30 min	Nanoplates, length of 151.6 nm and width of 14.3 nm	[78]
12	Template-assisted synthesis	Water	CeCl <sub>3</sub> ·7H <sub>2</sub> O, NaOH	SBA-15	Annealing at 700 °C	Mesoporous structure with pore-size centered at 3.5 nm	[79]
13	Template-assisted synthesis	Water	Ce(NO <sub>3</sub> ) <sub>3</sub> ·6H <sub>2</sub> O, NaOH	СТАВ	Annealing at 400 °C for 4 h	Mesoporous structure with pore-size centered at 9.5 nm	[80]
14	Solution-precipitation	Water	Cerium acetate hydrate, ammonia	ethylhexyl)	Precipitating at 67 °C for 72 h, followed by annealing at 400 °C for 4 h	Nanowires with diameters of 30 $-120\text{nm}$ and lengths of 0.2–5 $\mu\text{m}$	[81]
15	Potentiostatic electrochemical deposition	Ethyl alcohol	$CeCl_3 \cdot 7H_2O$	AAO	−10 V vs. SCE for different times	Nanotubes with diameters of 210 nm and lengths of $60  \mu m$	[82]
16	Solution-precipitation	Water	$Ce(NO_3)_3 \cdot 6H_2O$ , oxalic acid	Anodic alumina membranes	Precipitating for 3 h, followed by annealing at 700 °C for 10 h	Nanowires with diameters of 60 nm and	[83]
17	Solution-precipitation	Water	Ce(NO <sub>3</sub> ) <sub>3</sub> ·6H <sub>2</sub> O, HMT	PS microspheres	Precipitating at 75 °C for 2 h, followed by annealing at 600 °C for 2 h	Hollow microspheres with various shell thickness	[84]
18	Hydrothermal	Water	CeCl <sub>3</sub> ·6H <sub>2</sub> O, ammonia	Sodium oleate	$180^{\circ}\text{C}$ for 48 h, followed by annealing at 400 $^{\circ}\text{C}$ for 2 min	Ultrathin nanosheets with thickness of about 0.6 nm	[85]

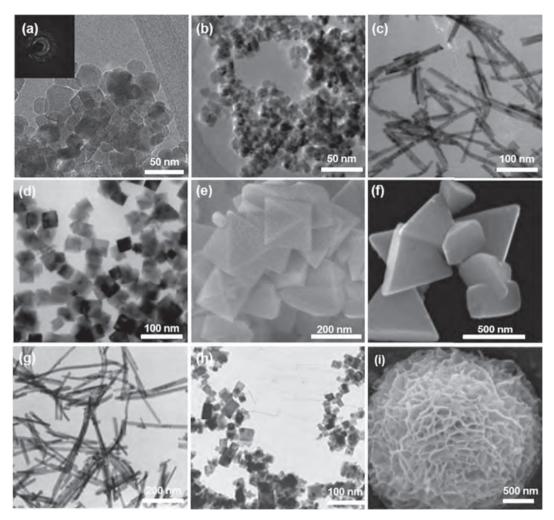
#### 3. Synthesis of CeO<sub>2</sub> catalysts

Nanosize effects on the catalysts' performance are generally observed in heterogeneous catalysis. In particular, when the dimension of catalytic materials decreases down to the range of 1–10 nm or even to subnanometers/single atom, strong size effects are observed to impose on their catalytic activity, selectivity, and stability because of the distinct changes in the surface physicochemical properties of catalysts as compared with their bulk counterparts [57–63]. Although the origin of these nanosize effects is still not thoroughly understood, many experimental studies have witnessed the importance of size controllability on the catalysts for various reactions. In principle, solid catalysts with the smaller sizes offer the higher catalytic activity and/or the better chemoselectivity. The morphology of catalysts has also been found to have

an equal importance in the modulation of their surface chemistry and the manipulation of electronic structures of their surface active sites. As a result, considerable attentions have been paid on the synthesis of nanocatalysts with tailorable sizes and shapes. In this section, the synthetic strategies of  $CeO_2$  nanocatalysts (Table 1) are given and thoroughly discussed: (1) hydrothermal synthesis, (2) solvothermal synthesis, and (3) template-assisted synthesis.

#### 3.1. Hydrothermal synthesis

Hydrothermal synthesis is a facile and cost-effective method for the preparation of crystalline  $CeO_2$  powders. In the early studies, ultrafine  $CeO_2$  powders were prepared hydrothermally in the presence of an excess amount of ammonia in a stainless steel autoclave at temperatures of  $150-200\,^{\circ}C$  [64]. Nanocrystalline



**Fig. 2.** Representative morphologies of CeO<sub>2</sub> synthesized by hydrothermal methods (a) TEM image of CeO<sub>2</sub> nanocrystalline particles, adapted with permission from Ref. [65]. Copyright © 2007, Elsevier B. V. All rights preserved. (b-d) TEM images of polyhedral CeO<sub>2</sub> nanoparticles, CeO<sub>2</sub> nanorods, and CeO<sub>2</sub> nanocubes, respectively, adapted with permission from Ref. [68]. Copyright © 2005, American Chemical Society. (e) SEM image of CeO<sub>2</sub> octahedra, adapted with permission from Ref. [69]. Copyright © 2011, Elsevier B. V. All rights preserved. (f) SEM image of CeO<sub>2</sub> triangular microplates, adapted with permission from Ref. [72]. Copyright © 2006, American Chemical Society. (g-h) TEM image of CeO<sub>2</sub> manocubes and CeO<sub>2</sub> truncated octahedra, adapted with permission from Ref. [73]. Copyright © 2008, American Chemical Society. (i) SEM image of monodisperse flowerlike CeO<sub>2</sub> microspheres, adapted with permission from Ref. [74]. Copyright © 2006, American Chemical Society.

particles of CeO2 (Fig. 2a) could also be obtained by a lowtemperature hydrothermal process at 250 °C, where cerium hydroxide and ceria acetate were used as precursors [65]. The entire hydrothermal process has been recognized as the well-known dissolution/recrystallization mechanism [66,67]. On the basis of this growth mechanism, Yan et al. reported a template-free hydrothermal method to synthesize CeO2 nanostructures with wellcontrolled morphologies as shown in Fig. 2 [68]. Typically, the hydrothermal temperatures and concentrations of bases play important roles in controlling the morphologies of CeO2 nanostructures. In particular, anisotropic Ce(OH)3 nuclei would be formed once the Ce<sup>3+</sup> ion precursors are added into an aqueous NaOH solution. At low concentrations of NaOH (e.g. < 1.0 M) and low hydrothermal temperature (e.g. 100 °C), the chemical potential driving force for the anisotropic growth of Ce(OH)3 nuclei is inadequate, thereby leading to the formation of polyhedral CeO<sub>2</sub> nanoparticles enclosed with (111) and (100) facets (Fig. 2b). When the NaOH concentration is increased, it accelerates the dissolution/ recrystallization rates and results in the anisotropic growth of Ce(OH)3 nuclei. For the further increase in the concentration of NaOH over 6.0 M, pure CeO<sub>2</sub> nanorods can be synthesized (Fig. 2c). Notably, at a hydrothermal temperature of  $180 \,^{\circ}\text{C}$ ,  $\text{Ce}(\text{OH})_3$  nuclei would become unstable in which they can be readily oxidized to  $\text{CeO}_2$ . In this case,  $\text{CeO}_2$  nanocubes can be produced at a high concentration of NaOH (6.0 M, Fig. 2d). On the other hand,  $\text{CeO}_2$  octahedra can be made using a similar process, where urea is used during the synthesis (Fig. 2e) [69]. They can also synthesized using  $\text{Ce}(\text{NO}_3)_3$  as the cerium source and  $\text{Na}_3\text{PO}_4$  as the mineralizer by a hydrothermal process at  $170 \,^{\circ}\text{C}$  [70,71].

Later, CeO<sub>2</sub> nanostructures with controllable sizes and morphologies, and tailorable surface properties can be synthesized by modified hydrothermal methods in the presence of surfactants or with optimized hydrothermal parameters (e.g., temperature, pressure). For example, single-crystal Ce(OH)CO<sub>3</sub> triangular microplates were successfully made in the presence of the surfactant cetyl-trimethyl ammonium bromide (CTAB) during the hydrothermal process [72]. Thermal treatments on the obtained Ce(OH)CO<sub>3</sub> precursor could further produce CeO<sub>2</sub> triangular microplates with single-crystal structures (Fig. 2f). Small inorganic anions could also be used to control the growth of CeO<sub>2</sub> nanocrystals. In this manner, NO<sub>3</sub> was observed to be effective in the synthesis of nanocubes, while halides and SO<sub>4</sub><sup>2</sup> were found to induce the assembly of CeO<sub>2</sub>

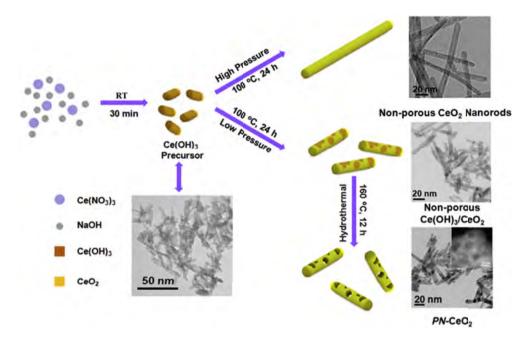
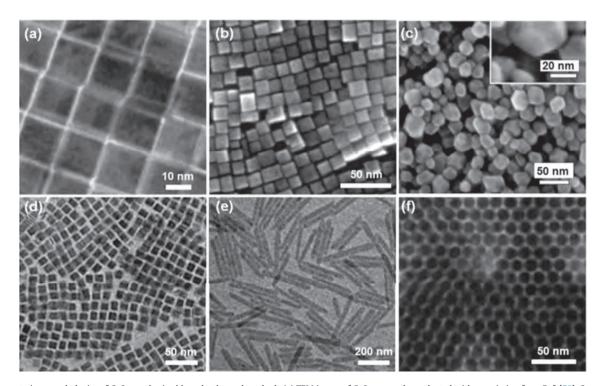


Fig. 3. Schematic illustration of the synthesis of PN-CeO<sub>2</sub> at low pressure. Its morphological evolution was revealed from TEM images. Adapted with permission from Ref. [37]. Copyright <sup>©</sup> 2014, Royal Society of Chemistry.



**Fig. 4.** Representative morphologies of CeO<sub>2</sub> synthesized by solvothermal methods (a) TEM image of CeO<sub>2</sub> nanocubes, adapted with permission from Ref. [76]. Copyright © 2006, American Chemical Society. (b) TEM image of CeO<sub>2</sub> nanocubes, adapted with permission from Ref. [77]. Copyright © 2012, Royal Society of Chemistry. (c) TEM image of CeO<sub>2</sub> truncated octahedra, adapted with permission from Ref. [77]. Copyright © 2012, Royal Society of Chemistry. (d—f) TEM images of square CeO<sub>2</sub> nanoplates, elongated CeO<sub>2</sub> nanoplates, and round CeO<sub>2</sub> nanoplates, respectively, adapted with permission from Ref. [78]. Copyright © 2012, WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim.

nanorods (Fig. 2g and h) [73]. Recently, nanostructured ceria could be hydrothermally prepared using the green deep eutectic solvent reline in the presence of SDS surfactants, which allows the facile morphology and porosity control in one of the less energy-intensive routes reported to date [75]. In the absence of water, only CeO<sub>2</sub> nanoparticles were obtained under any synthetic

temperatures. In contrast, the increased amount of water drove a morphological change of the as-synthesized CeO<sub>2</sub> nanostructures from nanoparticles to nanowires, suggesting that water was functioned as the directing agent [75]. Monodispersed flowerlike CeO<sub>2</sub> microspheres (Fig. 2i) could also be synthesized by the simultaneous polymerization-precipitation reaction, metamorphic

reconstruction, and mineralization under hydrothermal conditions followed by calcination [74]. The obtained CeO<sub>2</sub> microspheres usually consisted of 20- to 30-nm-thick nanosheets as the petals. Importantly, they were found to have the open three-dimensional (3D) porous and hollow structure, exhibiting the large surface area, large pore volume, and good thermal stability.

Recently, there is a surfactant-free hydrothermal method operated at low pressures which has been successfully developed to synthesize porous nanorods of  $CeO_2$  ( $PN-CeO_2$ ) [37]. The preparation involves a two-step hydrothermal process (Fig. 3). In the first step, a low pressure of ~1.2 atm is critical for the formation of a  $Ce(OH)_3/CeO_2$  nanorod precursor. Then, a subsequent hydrothermal step at a higher temperature ( $e.g. > 160\,^{\circ}C$ ) would lead to the complete dehydration and oxidation of  $Ce(OH)_3$  in the precursor nanorods and induce formation of porous nanorods successively. The unaltered rod-like morphology may be attributed to the robustness of precursor nanorods surviving through the dehydration, oxidation, and accompanied structure reorganization in the hydrothermal treatment.

#### 3.2. Solvothermal synthesis in the presence of capping agents

Despite the simplification of the hydrothermal method, its controllability on the geometries and morphologies of  $CeO_2$  nanostructures is still insufficient for practical utilizations. Thus, the solvothermal synthesis in the presence of various capping agents has been developed in order to realize this purpose. For example, a shape- and size-controllable synthesis of  $CeO_2$ 

nanocubes enclosed by six (200) planes was achieved by a rationally designed one-pot approach (Fig. 4a) [76]. The 2D and 3D selfassembly of CeO2 nanocubes was formed by the oriented aggregation-mediated precursor growth on the substrates. Both the shape and size of CeO2 nanostructures could then be tuned conveniently by adjusting the concentrations of reactants and stabilizing agents (i.e. oleic acid) as well as the water/toluene molar ratios. Also, Wang et al. was able to synthesize CeO2 nanocubes and CeO<sub>2</sub> truncated octahedra in the presence of various capping agents in an alkaline environment [77]. These nanocubes were made in the presence of oleic acid as the capping agent and using Ce(N- $O_3)_3 \cdot 6H_2O$  as the Ce source in an alkaline solution of tetramethylammonium hydroxide (TMAH) (Fig. 4b). The side lengths of the assynthesized nanocubes enclosed with six (100) facets are in the range of 9-17 nm. Similarly, CeO<sub>2</sub> truncated octahedra were prepared using polyvinyl pyrrolidone (PVP) as the capping agent in the reaction system (Fig. 4c). The size of CeO<sub>2</sub> truncated octahedra surrounded with eight (111) and six (100) facets was ~17 nm. Also, ultrathin 2D CeO<sub>2</sub> nanoplates were successfully prepared by the thermal decomposition of cerium acetate at 320-330 °C in the presence of oleic acid and oleylamine as the capping agent and the surfactant, respectively (Fig. 4d-f) [59]. The morphology of nanoplates can be easily controlled by the reaction parameters (e.g., precursor ratio, reaction time, and temperature).

#### 3.3. Template-assisted synthesis

The size, surface area, and shape of CeO<sub>2</sub> nanostructures can also

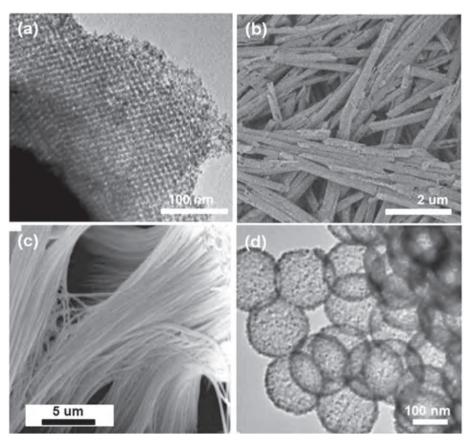


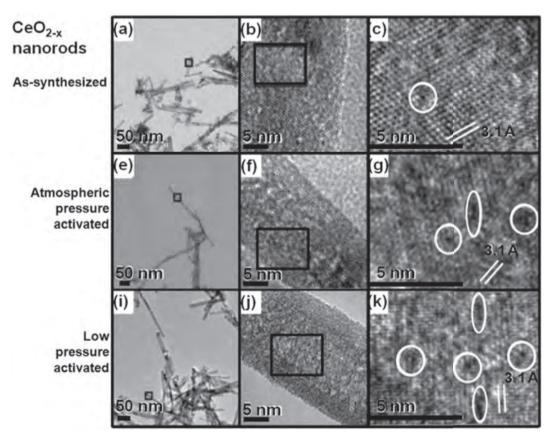
Fig. 5. Representative morphologies of CeO<sub>2</sub> synthesized by template methods (a) TEM image of the template-free CeO<sub>2</sub> samples, adapted with permission from Ref. [79]. Copyright © 2003, Royal Society of Chemistry. (b) SEM image of CeO<sub>2</sub> nanotubes, adapted with permission from Ref. [82]. Copyright © 2009, American Chemical Society. (c) SEM image of CeO<sub>2</sub> nanowire arrays, adapted with permission from Ref. [83]. Copyright © 2004, Elsevier B. V. All rights preserved. (d) TEM image of porous hollow CeO<sub>2</sub> microspheres, adapted with permission from Ref. [84]. Copyright © 2012, Springer International Publishing AG.

**Table 2**Summary of surface control methods.

# Method	Conditions	Effect on surface structure	application	Ref.
1 High temperature annealing under vacuum	Annealing in vacuum at different temperatures	Increasing lattice defect and oxygen storage capacity (OSC)	-	[86]
2 Hydrogen treatment	Annealing at 700, 800 and 900 °C with mixture gas of nitrogen and hydrogen for 2 h, respectively	Increased surface Ce <sup>3+</sup> content and concentrations of oxygen vacancy, modulated optical property	Up- and down- conversion	[87]
3 Thermal activation in a gas mixture of nitrogen and oxygen	Annealing at 400 $^{\circ}\text{C}$ with nitrogen-oxygen mixture gas under vacuum of 0.1 Torr	Changes in the surface $Ce^{3+}$ fractions of the nanostructured $CeO_2$ and in the coordination numbers of the surface cerium atoms	CO oxidation	[88]
4 In situ and ex situ thermal annealing treatments	Annealing at 375 and 800 $^{\circ}\text{C}$ in an oxidizing environment	Disordered small oxygen defects and small vacancy clusters, increased lattice strain and expansion, formation of polyhedral nanocavities	_	[89]
5 EB irradiation	EB energy of 0.7 MeV and an initial EB current of 2 mA/s in water at room temperature and atmospheric pressure.	More surface Ce <sup>3+</sup> species and increased the concentration of surface defects	Photocatalysis	[90]
6 Wet chemical redox etching	Treating with ascorbic acid and hydrogen peroxide alternatively for different cycles	Larger surface area, more surface Ce <sup>3+</sup> fraction, higher concentration of oxygen vacancy	CO oxidation	[92]
7 Pressure control	Hydrothermal under different pressures and oxygen/ nitrogen ratios	Controllable length/diameter ratio, surface area, surface $\mathrm{Ce}^{3+}$ fraction and OSC value	CO oxidation and nitrile hydrolysis	[93]

be modulated by using templates during the synthesis such as organic macromolecules, surfactants, nanometer-sized granules of silica, polycarbonate membranes, and anodic alumina (AAO) membranes. Among all, mesoporous silica templates have been successfully employed to fabricate the highly ordered mesoporous CeO<sub>2</sub> with uniform mesopore sizes [79]. The synthesis involves the incorporation of cerium precursors into a mesoporous silica

template, subsequent conversion of precursors into CeO<sub>2</sub>, and final removal of the silica template. The morphology of obtained mesoporous CeO<sub>2</sub> is very similar to that of silica templates (Fig. 5a). Importantly, these mesoporous CeO<sub>2</sub> are found to exhibit the remarkable thermal stability at 973 K. Also, nanoscaled CeO<sub>2</sub> with a large surface area of  $200 \, \mathrm{m}^2 \, \mathrm{g}^{-1}$  could be prepared using cetyl-trimethyl ammonium bromide (CTAB) as the templating agent [80],



**Fig. 6.** Surface properties of CeO<sub>2</sub> modulated by the thermal activation method in a gas mixture of nitrogen and oxygen at various pressures. (a–c) TEM images of the assynthesized CeO<sub>2</sub> nanorods; (e–g) TEM images of the thermally activated CeO<sub>2</sub> nanorods at atmospheric pressure; and (i–k) TEM images of the thermally activated CeO<sub>2</sub> nanorods at low pressure (0.1 Torr). The left column shows the typical TEM images of CeO<sub>2</sub> nanorods. Squares in the figures show approximate zoomed-in area of interest for adjacent figures in the right column. The middle column shows HRTEM images of individual nanorods. The right column illustrates the approximate locations of some oxygen vacancy defects: point oxygen vacancies (circles) and linear oxygen vacancies (ovals). Adapted with permission from Ref. [87]. Copyright © 2011, American Chemical Society.

**Table 3** Parameters of the local structure around Ce atoms of low pressure-activated ceria nanorods, nanoparticles, and bulk samples obtained from the curve fitting of Ce  $L_1$ -edge EXAFS. Adapted with permission from Ref. [88]. Copyright  $^{\circ}$  2011, American Chemical Society.

CeO <sub>2-x</sub> sample	neighboring atom	N	R(Å)
theoretical	0	8	2.34
	Ce	12	3.83
	O	24	4.48
nanorod	O	$6.3 \pm 0.1$	$2.294 \pm 0.005$
	Ce	$7.4 \pm 0.4$	$3.815 \pm 0.005$
	O	$14.7 \pm 1.2$	$4.515 \pm 0.005$
nanoparticle	O	$6.5 \pm 0.1$	$2.299 \pm 0.005$
	Ce	$6.5 \pm 0.3$	$3.841 \pm 0.005$
	O	$15.5 \pm 1.3$	$4.464 \pm 0.005$
bulk	O	$8 \pm 0.1$	$2.316 \pm 0.005$
	Ce	$12 \pm 0.3$	$3.82 \pm 0.005$
	O	$24 \pm 1.2$	$4.48 \pm 0.005$

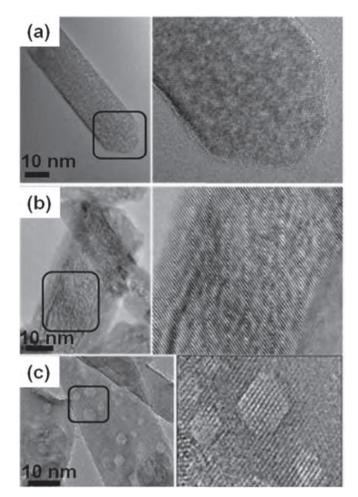
 $^aN$  is the coordination number. R is the average interatomic distance between Ce and its neighboring atoms. Uncertainties were estimated by the double-minimum residue  $2\chi^2$  method.

while polycrystalline CeO2 nanowires were synthesized via a solution-phase route with sodium bis(2-ethylhexyl) sulfosuccinate as the templating agent [81]. It is noted that the as-synthesized nanowires were composed of many tiny grains at different orientations. At the same time, CeO<sub>2</sub> nanotube arrays (Fig. 5b) could as well be successfully fabricated by a potentiostatic electrochemical deposition from a cerium nitride solution, where the AAO membranes were served as the hard templates [82]. As expected, these CeO<sub>2</sub> nanotubes showed a polycrystalline structure with a thin wall of 6 nm and uniform diameter. In addition, using the similar AAO membranes as templates, ordered CeO<sub>2</sub> nanowire arrays (Fig. 5c) could also be prepared [83]. During the deposition process, Ce<sup>3+</sup> and  $C_2O_4^{2-}$  species conversely migrated into the hexagonally ordered nanochannels of AAO membranes and reacted inside the channels to form Ce<sub>2</sub>(C<sub>2</sub>O<sub>4</sub>)<sub>3</sub> precursors configured into the onedimensional nanowire geometry along the axial direction. The subsequent thermal annealing of Ce<sub>2</sub>(C<sub>2</sub>O<sub>4</sub>)<sub>3</sub>/AAO would result in the formation of CeO2 nanowires confined in the pores of AAO templates. On the other hand, porous hollow CeO2 microspheres (Fig. 5d) with the good monodispersion and uniformity in size were synthesized, in which the negatively charged polystyrene (PS) microspheres were used as the hard templates [84], whereas ultrathin CeO<sub>2</sub> sheets were also synthesized from the hard template of ultrathin CeCO<sub>3</sub>OH sheet precursors by a well-controllable thermal annealing process in air [85]. All these examples demonstrate the versatility of different template-assisted methods for the synthesis of various CeO2 nanostructures.

#### 4. Surface control on CeO<sub>2</sub> nanomaterials

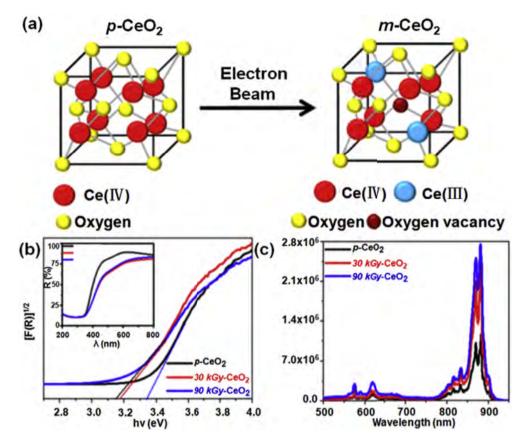
In general, the synthetic method would dictate the surface properties of obtained CeO<sub>2</sub> nanostructures, including their specific surface area, surface Ce<sup>3+</sup> fraction, and oxygen vacancy concentration. During the formation of CeO<sub>2</sub> and their related precursors, various strategies of doping foreign elements as well as controlling the morphologies and the exposed facets of these nanostructures are also extensively studied. For example, different post-synthesis treatments, including the high-temperature annealing processes under selected temperatures, gas atmospheres and pressures, and electron beams (EBs) and chemical based modifications have been widely developed in order to regulate the surface properties of the nanostructures [86–88]. Table 2 summarizes the recent efforts on regulating the surface properties of CeO<sub>2</sub>.

Mamontov et al. demonstrated the dependence of lattice defect

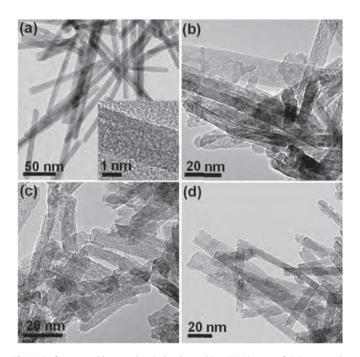


**Fig. 7.** Surface control on  $CeO_2$  nanorods by thermal annealing. *Ex situ* annealing: HRTEM images showing various amounts of imperfect crystallinity (image pairs with zoom-in of black rectangles): (a) As-synthesized  $CeO_2$  nanorods: Rod direction <110>. (b)  $CeO_2$  nanorods annealed at 325 °C in air. (c)  $CeO_2$  nanorods annealed at 800 °C: rod direction for (b, c) is <211>. The viewing directions for (a-c) are of <110> type. Adapted with permission from Ref. [89]. Copyright ° 2015, Royal Society of Chemistry.

formation and OSC on the annealing temperature under vacuum, suggesting that the structural defect concentration is maximized at medium annealing temperatures of 300-600°C [86]. Also, hydrogen treatments on the erbium-doped CeO2 nanoparticles were observed to lead to the chemical reduction of Ce<sup>4+</sup> as well as the increase in the surface Ce<sup>3+</sup> content and the concentration of oxygen vacancy [87]. Thus, they exhibited enhanced optical properties with the higher up-conversion and down-conversion efficiencies. The manipulation on the oxygen vacancies of CeO<sub>2</sub> nanoparticles and nanorods could also be realized by the thermal activation method in a gas mixture of nitrogen and oxygen at a low pressure of 0.1 Torr and atmospheric pressure, respectively (Fig. 6). All these activations would lead to the changes in surface Ce<sup>3+</sup> fractions of the nanostructured CeO2 and in coordination numbers of the surface cerium atoms (Table 3) [88]. Moreover, the process pressure was known to considerably affect the surface properties of fabricated CeO<sub>2</sub>. Low-pressure treatments produced CeO<sub>2</sub> with a higher surface Ce<sup>3+</sup> fraction, while those under atmospheric pressure resulted in a relatively lower surface Ce<sup>3+</sup> fraction but still higher than that of the untreated CeO2. Thus, the enhanced catalytic activity for CO oxidation reaction was observed for the CeO<sub>2</sub> based catalysts treated at various pressures, which may be attributed to the increase in surface defects in terms of higher surface



**Fig. 8.** Surface properties of CeO<sub>2</sub> controlled by EB irradiation. (a) Schematic model illustrating the change in crystalline structure of CeO<sub>2</sub> due to the formation of Ce(III) and oxygen vacancy after EB irradiation. (b)  $[F(R)h\nu]^{1/2}$  versus  $h\nu$  plot of p-CeO<sub>2</sub>, 30 kGy-CeO<sub>2</sub>, and 90 kGy-CeO<sub>2</sub> nanostructures. (lnset) UV—vis diffuse reflectance spectra. (c) PL spectra of p-CeO<sub>2</sub>, 30 kGy-CeO<sub>2</sub>, and 90 kGy-CeO<sub>2</sub> nanostructures. Adapted with permission from Ref. [90]. Copyright  $^{\odot}$  2014, American Chemical Society.



**Fig. 9.** Surface control by wet chemical redox etching. TEM images of  $CeO_2$  nanorods. (a) As-synthesized  $CeO_2$  nanorods (inset: HRTEM image of  $CeO_2$  nanorods). (b)  $CeO_2$  nanorods after one cycle, (c) four cycles, and (d) eight cycles of oxidation/reduction treatment. Adapted with permission from Ref. [92]. Copyright  $^{\odot}$  2015, Royal Society of Chemistry.

#### Ce<sup>3+</sup> fractions.

Recently, the tailorable structures and surface properties of  $CeO_2$  nanorods were also achieved by the *in situ* and *ex situ* thermal annealing treatments (Fig. 7) [89]. Two different types of oxygen-related defects were found under TEM observations: (1) the disordered oxygen defects and (2) the vacancy clusters. Both types of defects would increase the lattice strain and the expansion of assynthesized  $CeO_2$  nanorods. In addition, the thermal treatments would lead to the formation of polyhedral nanocavities within the nanorods due to the oxygen migration within the surface defect clusters. Molecular dynamic simulation also confirmed the agglomeration of oxygen vacancies within the as-prepared nanorods, which was ascribed to the evolution of nanocavities on the surface of  $CeO_2$  nanorods.

However, the commonly used high-temperature treatment approach, somehow, is destructive because of the fragile  $\text{CeO}_2$  nanostructures at high temperatures, which leads to the apparent sintering and consequent loss of specific surface areas as well as shielding/blocking of the surface active sites. Several novel strategies, including various physical and chemical methods, were then successfully developed to regulate the surface properties of  $\text{CeO}_2$  with the aim to modulate its catalytic activity and selectivity.

The EB irradiation method was then developed to create more surface defects on the  $CeO_2$  surface [90]. Various doses of EB irradiation received on  $CeO_2$  nanoparticles could change their surface chemical states and compositions. A high dose generated more surface  $Ce^{3+}$  species and increased the concentration of the surface defects (Fig. 8). The defect-engineered  $CeO_2$  nanoparticles would then facilitate their photocatalytic activity under visible light by enhancing the electron transfer and suppressing the recombination

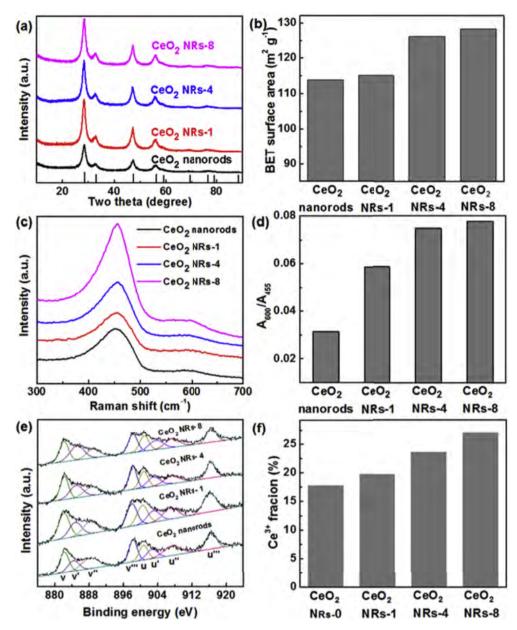


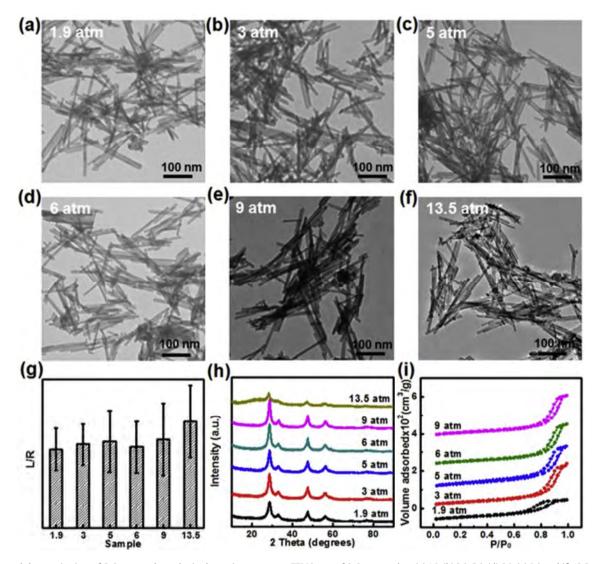
Fig. 10. Surface properties of CeO<sub>2</sub> nanorods treated with various cycles of the wet chemical etching. (a) XRD patterns, (b) Specific surface areas, and (c) Raman spectra of the assynthesized CeO<sub>2</sub> nanorods and the treated CeO<sub>2</sub> nanorods with various cycles (1, 4, and 8) of the chemical reduction/oxidation. (d) Area ratio of Raman peaks at 455 and 600 cm<sup>-1</sup> for all CeO<sub>2</sub> nanorods derived from their Raman spectroscopic profiles, as shown in (c). (e) XPS spectra of Ce 3d core level regions, and (f) Surface Ce<sup>3+</sup> fractions of the assynthesized CeO<sub>2</sub> nanorods and the etched CeO<sub>2</sub> nanorods with 1, 4, and 8 reduction/oxidation cycles. Adapted with permission from Ref. [92]. Copyright © 2015, Royal Society of Chemistry.

of photogenerated holes and electrons.

Also, the post-treatment induced modulation of surface properties of CeO<sub>2</sub> and their morphology modification could be realized by the wet chemical redox etching approach [91]. Recently, our group demonstrated a facile wet chemical method to engineer the surface properties of CeO<sub>2</sub> nanorods [92]. In this work, ascorbic acid (AA) and hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>) were functioned as the reducing agent and oxidizing agent, respectively, to react with the surface Ce<sup>3+</sup> and Ce<sup>4+</sup> species of CeO<sub>2</sub> nanorods. The processed CeO<sub>2</sub> nanorods using AA and H<sub>2</sub>O<sub>2</sub> alternatively for various redox etching cycles could not only deliver the morphology evolution from a smooth surfaced to a rough surfaced structure but also give the controllable surface properties there (Fig. 9).

After the chemical etching process, the fluorite crystal structure and the rod-like morphology of CeO<sub>2</sub> nanorods were preserved,

similar to that of the as-synthesized CeO<sub>2</sub> nanorods (Fig. 10a). Importantly, the surface properties of CeO<sub>2</sub> can be effectively controlled by the number of chemical etching cycles. The specific surface areas of etched nanorods were increased to 115.1, 126.4, and 128.3 m<sup>2</sup> g<sup>-1</sup> after 1, 4, and 8 cycles of the treatments (Fig. 10b), respectively. Raman spectra also illustrated that a higher number of chemical reduction/oxidation cycles would lead to an increase in concentration of surface oxygen vacancies (Fig. 10c and d). The surface Ce<sup>3+</sup> fractions of the processed CeO<sub>2</sub> nanorods was found to increase steadily with the increasing number of chemical redox cycles (Fig. 10e), which was consistent with the morphological changes observed under TEM and the evolutions of the measured peaks in their Raman spectra (Fig. 10f). Hence, the chemical etching process is known to be a facile approach to manipulate the controllable surface properties of CeO<sub>2</sub> for the desired applications.



**Fig. 11.** Structural characterizations of CeO<sub>2</sub> nanorods synthesized at various pressures. TEM image of CeO<sub>2</sub> nanorods at (a) 1.9, (b) 3.0, 5.0, (d) 6.0, (e) 9.0, and (f) 13.5 atm. (g) Plot of the L/W ratios of CeO<sub>2</sub> nanorods vs various pressures of the reaction vessels. (h) XRD patterns of CeO<sub>2</sub> nanorods. (i) N<sub>2</sub> adsorption-desorption isotherms of CeO<sub>2</sub> nanorods. Adapted with permission from Ref. [93]. Copyright <sup>©</sup> 2016, American Chemical Society.

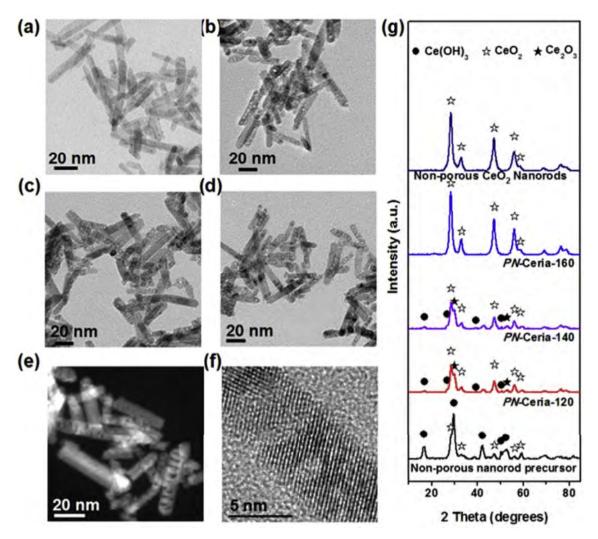
**Table 4** Structural characteristics of  $CeO_2$  nanorods synthesized at various pressures. Adapted with permission from Ref. [93]. Copyright  $^{\circ}$  2016, American Chemical Society.

Pressure (atm)	Ce <sup>3+</sup> (%)	BET (m <sup>2</sup> g <sup>-1</sup> )	OSC (mmol O <sub>2</sub> per g)
1.9	17.6	93.4	167.9
3	21.5	106.2	211.9
5	23.4	105.6	280.8
6	22.3	98.9	238.1
9	17.7	96.8	183.9
13.5	16.4	92.7	181.2

Another effective strategy to controllably adjust the surface properties of CeO<sub>2</sub> nanorods is to regulate the synthetic pressures and the partial pressure of oxygen in reaction vessels during the hydrothermal processes [37,93]. Neither the air pressures nor the oxygen partial pressures would alter the rod-like morphology of CeO<sub>2</sub> except the different length/diameter ratio (Fig. 11). However, their surface properties, including the surface Ce<sup>3+</sup> fractions, surface areas, and OSC values, could be effectively modulated by the total pressure and partial pressure of oxygen as summarized in

Table 4. An optimized pressure of 5.0 atm with a relatively low oxygen partial pressure was observed to produce  $CeO_2$  nanorods with the highest concentration of oxygen vacancy, highest surface fraction of  $Ce^{3+}$  species, and largest value of OSC [92]. Generally, the growth mechanism of  $CeO_2$  nanorods is considered as the formation of anisotropic rod-like  $Ce(OH)_3$  nuclei followed by the subsequent dissolution/recrystallization process of the nuclei [70,71]. Thus, the dissolution/recrystallization rate of  $Ce(OH)_3$  and the solubility of  $O_2$  in the solution under various pressures essentially determine the surface properties of  $CeO_2$  nanorods. These two factors would make  $CeO_2$  nanorods reaching their maximum concentration of the surface defects at 5.0 atm. Therefore, the pressure is believed to play a considerable role in tailoring the surface  $Ce^{3+}$  fraction and the surface-bound defects of  $CeO_2$  nanorods.

When the synthetic pressure was inversely reduced to ~1.2 atm, the hydrothermal approach would produce the rod-like precursor, composed of 41.5%  $Ce(OH)_3$  and 58.5% cubic fluorite  $CeO_2$ , with a diameter of ~8 nm and a length of ~60 nm, as illustrated in Fig. 12. The second hydrothermal treatment over 160 °C would then lead to a new form of PN- $CeO_2$ , which has a very large surface area of 144 m<sup>2</sup>/g and a high surface  $Ce^{3+}$  fraction of 32.8%. Impressively,



**Fig. 12.** Structural characterizations and surface properties of *PN*-CeO<sub>2</sub> (a) TEM image of the nonporous nanorod precursor. (b) TEM image of *PN*-CeO<sub>2</sub>-120. (c) TEM image of *PN*-CeO<sub>2</sub>-140. (d) TEM image of *PN*-CeO<sub>2</sub>-160. (e) Dark-field TEM image of *PN*-CeO<sub>2</sub>-160. (f) HRTEM image of *PN*-CeO<sub>2</sub>-160. (g) XRD patterns of ceria-based nanostructures. Adapted with permission from Ref. [36]. Copyright <sup>©</sup> 2014, Royal Society of Chemistry.

**Table 5**Structural characteristics of nanoceria synthesized at various conditions. Adapted with permission from Ref. [37]. Copyright © 2014, Royal Society of Chemistry.

Ce <sup>3+</sup> fraction (%)	$BET(m^2 g^{-1})$	OSC (mmol O <sub>2</sub> per g)
19.1	109	_
24.3	120	_
30.8	141	900.2
21.6	137	831.2
14.5	107	167.9
16.7	12.2	84.2
19.0	8.4	152.7
	19.1 24.3 30.8 21.6 14.5 16.7	19.1 109 24.3 120 30.8 141 21.6 137 14.5 107 16.7 12.2

*PN*-CeO<sub>2</sub> also delivers an extremely large OSC value up to ~900 μmol O<sub>2</sub>/g, being over four times higher than the values reported for any other nanostructured forms of ceria (Table 5). Furthermore, the surface properties of *PN*-CeO<sub>2</sub> can also be regulated by the hydrothermal temperatures at the second step. Its morphological features can as well be well preserved. These porous nanorods could reach the similar surface areas of 141, 131, and  $122 \, \mathrm{m}^2/\mathrm{g}$  for the catalysts synthesized at 160, 180, and  $200 \, ^{\circ}\mathrm{C}$ , respectively. The surface Ce<sup>3+</sup> fractions and the concentrations of oxygen vacancy would generally decrease with the increasing

hydrothermal temperature during the reaction.

### 5. Catalytic performance

#### 5.1. CO oxidation

Removal of carbon monoxide (CO) from a specific gas atmosphere is of great importance in daily life and industrial applications. For example, CO in the exhaust gas emission from vehicles due to the incomplete combustion of gasoline pollutes the environment and threatens the human health directly. Also, industrial production of hydrogen is usually done by the thermal reformation of coal, hydrocarbons or alcohols via the following reaction: fuels  $+ O_2 + H_2O \rightarrow CO_x + H_2$ . This reforming process always produces CO and  $CO_2$ . The water-gas shift (WGS) reaction (CO  $+ H_2O \rightarrow CO_2 + H_2$ ) is then used to remove most of the accompanied CO [94]. However, ~1% of CO remains in the typical WGS effluent gas, which is a negative factor for any further catalytic applications. The CO residue can poison majority of the metal-based catalysts. Thus, it is necessary and important to remove CO to a trace level below 10 ppm. In this regard, CeO<sub>2</sub> is considered as a promising candidate for CO oxidation as both additive and catalytic

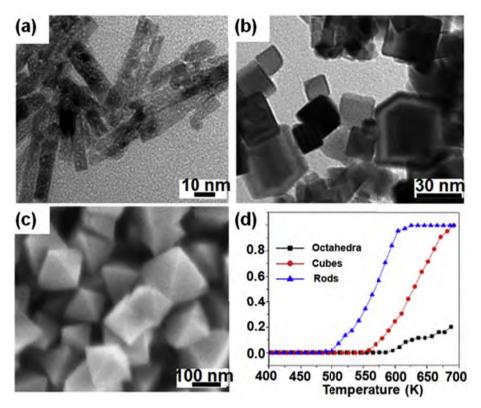


Fig. 13. (a) TEM image of ceria rods, (b) TEM image of cubes, and (c) SEM image of octahedra after CO oxidation to 673 K, and (d) light-off curves for CO oxidation over ceria rods, cubes, and octahedra. Adapted with permission from Ref. [104]. Copyright © 2012, Elsevier B. V. All rights preserved.

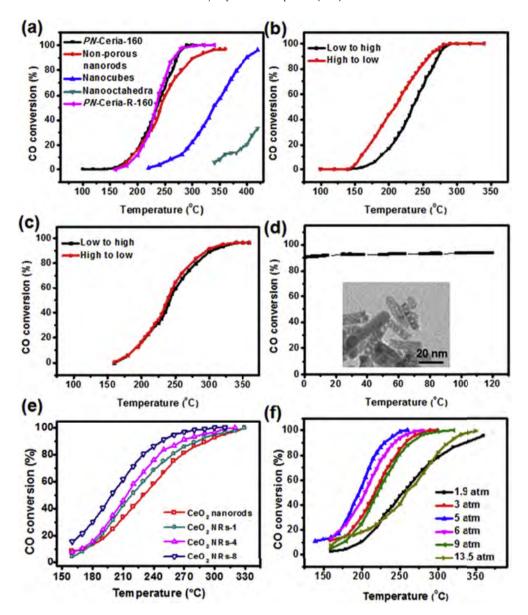
active component because of its reversible redox chemistry, large OSC value, and good thermal durability. Although many studies have demonstrated the excellent catalytic properties of CeO<sub>2</sub>-incooperated noble metals (Pt, Au, and Pd, etc.), metal oxides (CuO, and Co<sub>3</sub>O<sub>4</sub>, etc.), CeO<sub>2</sub> and transition metal doped CeO<sub>2</sub> catalysts for CO oxidation as well as preferential CO oxidation in the presence of excessive H<sub>2</sub> have been rarely studied [32,94–103]. Herein, we emphasize on the relationship between CO oxidation activity and surface properties of pure CeO<sub>2</sub> catalysts.

CO oxidation reaction over CeO<sub>2</sub> usually precedes the Mars-van Krevelen mechanism, including the adsorption of CO on the top of surface Ce<sup>3+</sup>, the activation of CO by the lattice oxygen to form the intermediate COO\*, the removal of CO<sub>2</sub> accompanied by the formation of oxygen vacancies, and the consequent annihilation of oxygen vacancies by gas phase oxygen [104]. Thus, the catalytic activity of CeO<sub>2</sub> for CO oxidation is well determined by its surface properties. In general, surface properties, such as the specific surface areas, surface Ce<sup>3+</sup> fractions, surface planes, and oxygen vacancies are considered as the main factors to affect the performance of CeO<sub>2</sub> towards CO oxidation. CeO<sub>2</sub> catalysts with the features of high surface area, large concentration of surface defects, high mobility of lattice oxygen and low formation energy of oxygen vacancy can benefit CO oxidation. In this case, many efforts have been focused on the surface control of CeO<sub>2</sub>.

Recently, several individual groups reported the significant effect of the exposed crystal planes of differently shaped CeO<sub>2</sub> catalysts on their CO oxidation activity, such as nanowires, nanorods, nanocubes, nanooctahedra, nanospindles, nanospheres, and nanotubes [37,104–107] Overbury et al. studied the shape-activity relationships of various CeO<sub>2</sub> nanostructures for CO oxidation by using *in situ* diffuse reflectance infrared Fourier transform spectroscopy (DRIFTS) techniques (Fig. 13) [104]. The reactivity for CO oxidation is strongly surface dependent with the following trend:

nanorods > nanocubes > octahedral. The difference among surface oxygen vacancy formation energy, amount of low coordination sites and defects sites in various crystal planes of the differently shaped CeO<sub>2</sub> catalysts are recognized as the driving force of the surfacedependent catalytic behaviors for CO oxidation. The low oxygen vacancy formation energy and high mobility of lattice oxygen in CeO<sub>2</sub> (110) crystal plane (i.e. the dominated plane of CeO<sub>2</sub> nanorods) as well as the nature and amount of defects and low coordination sites of CeO<sub>2</sub> nanorods would enable the highest catalytic activity of CeO<sub>2</sub> nanorods for CO oxidation [102]. For the same reason, the similar surface-dependent catalytic phenomenon for Deacon process (i.e. a gas phase oxidation for converting HCl into Cl<sub>2</sub>) catalyzed by CeO<sub>2</sub> was also observed [108-110]. Among CeO<sub>2</sub> catalysts with various morphologies, CeO2 nanorods could deliver the highest catalytic activity and the best stability for Deacon oxidation [110].

To date, our group reported a new type of surface defectabundant CeO<sub>2</sub> nanorods (PN-CeO<sub>2</sub>) fabricated by a two-step hydrothermal method, which showed a very high catalytic activity for CO oxidation [37]. As shown in the catalytic light-off curve (Fig. 14), these PN-CeO2 nanorods delivered the significantly improved CO oxidation activity and durability as compared with other CeO2 catalysts. This improved CO oxidation performance of PN-CeO<sub>2</sub> can be attributed to the richer surface defects and the larger OSC value of the catalysts with abundant surface Ce<sup>3+</sup> species as well as oxygen vacancies. PN-CeO2 also showed a catalytic activitytemperature hysteresis between the light-off measurement spanning from low to high temperatures and the subsequent measurement covering from high to low temperatures (Fig. 14c). Such hysteresis can be rationalized in terms of the "overheating of active sites of the catalysts when decreasing the temperature of the reactor chamber." The broad width (25 °C) of the hysteresis at 50% of CO conversion catalyzed by PN-CeO<sub>2</sub> was much larger than that



**Fig. 14.** CO oxidation performance of CeO<sub>2</sub> nanostructures. (a) CO oxidation catalyzed by various CeO<sub>2</sub> catalysts. (b) Temperature hysteresis of CO conversion on *PN*-CeO<sub>2</sub>. (c) Temperature hysteresis of CO conversion catalyzed by nonporous CeO<sub>2</sub> nanorods. (d) Stability of *PN*-CeO<sub>2</sub> for CO oxidation at 270 °C. The inset is the TEM image of *PN*-CeO<sub>2</sub> after a 120-h stream-on-line running. Adapted with permission from Ref. [37]. Copyright © 2014, Royal Society of Chemistry. (e) CO oxidation catalyzed by CeO<sub>2</sub> nanorods with various chemical etching cycles. Adapted with permission from Ref. [92]. Copyright © 2015, Royal Society of Chemistry. (f) CO oxidation catalyzed by CeO<sub>2</sub> nanorods prepared under various pressures. Adapted with permission from Ref. [93]. Copyright © 2016, American Chemical Society.

 $(4\,^{\circ}\text{C})$  by nonporous  $\text{CeO}_2$  nanorods, reflecting the high CO oxidation activity of  $PN\text{-CeO}_2$  with a larger width being associated with a more active catalyst. A similar catalytic performance of the surface-engineered  $\text{CeO}_2$  for CO oxidation was also observed in the pressure-regulated  $\text{CeO}_2$  nanorods. All results presented here indicate the strong correlation among the high surface area, large number of structural defects, and high faction of  $\text{Ce}^{3+}$  of  $\text{CeO}_2$  catalysts and their catalytic performance for CO oxidation.

Furthermore, the catalytic performance of CeO<sub>2</sub> nanorods regulated by the redox chemical etching methods [92] or pressure control [93] was also observed with the same tendency. As shown in Fig. 14e, the processed CeO<sub>2</sub> nanorods with the increased etching cycles would induce a steady increase in their surface areas, oxygen vacancies and surface Ce<sup>3+</sup> fractions, which delivered the higher catalytic activity for CO oxidation. By optimizing the synthetic pressures, the CeO<sub>2</sub> nanorods prepared at 5.0 atm would present

the enhanced surface Ce<sup>3+</sup> fractions and the larger concentration of oxygen vacancy, leading to the superior catalytic performance for CO oxidation (Fig. 14f).

Lately, atomically thin CeO<sub>2</sub> nanosheets with the abundant surface pits were exhibited with a very high catalytic activity for CO oxidation (Fig. 15). These ultrathin catalysts displayed the structural features of the existence of many pits (Fig. 15a). X-ray absorption spectroscopy revealed the fine structure of the nanosheets, in which there were many pits surrounding surface Ce sites with a very low average coordination number of 4.6. Fine structural characterizations showed that the coordination numbers of cerium species surrounded by the pits were mainly 4, 5, and 6 (Fig. 15d). These sites are widely recognized as the catalytically active sites for CO oxidation [85]. In this unique configuration, the CO molecules are preferentially adsorbed on top of the four-coordinated pits surrounding Ce sites, while the O<sub>2</sub> molecules yield a strong

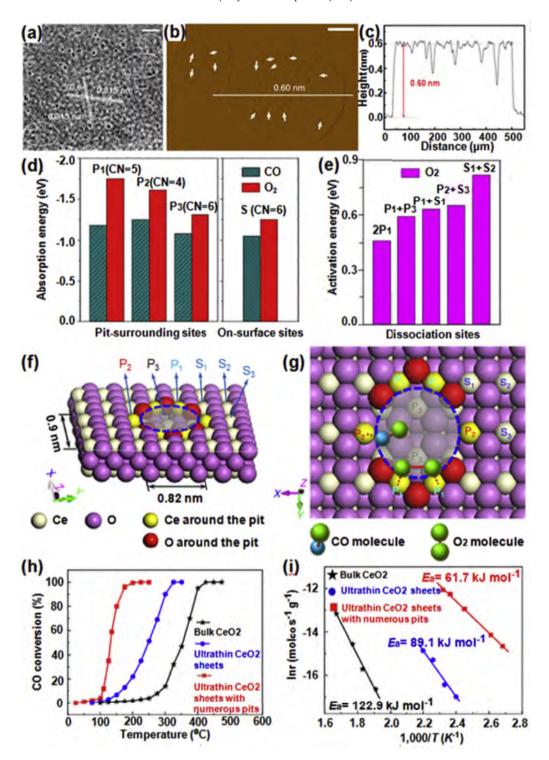


Fig. 15.  $CeO_2$  nanosheets with abundant surface pits for CO oxidation and their catalytic mechanism. (a) HRTEM image, (b) AFM image, and (c, d) Height profiles along the white line indicated in (b) for the three-atom-thin  $CeO_2$  sheets with numerous surface pits. The scale bars in (c) and (d) are 3 and 100 nm, respectively. The inset circles and arrows in (a) and (b) denote the presence of pits on their surface. (d) Calculated adsorption energies for CO and  $O_2$  molecules on the pits surrounding Ce sites  $(P_1-P_2)$  and on surface Ce sites  $(S_1-S_3)$ , and (e) Calculated activation energies for  $O_2$  dissociation on adjacent Ce sites of ultrathin  $CeO_2$  sheets with numerous pits; the coordination numbers of  $P_1$ ,  $P_2$ ,  $P_3$ ,  $S_1$ ,  $S_2$ , and  $S_3$  are  $S_3$ ,  $S_4$ ,

adsorption at the five-coordinated pits surrounding Ce species (Fig. 15c). Such configuration of the catalytic centers can avoid the catalyst poison and promote the catalytic activity for CO oxidation (Fig. 15d-f). DFT calculations also showed that the dissociation of O<sub>2</sub> is the rate-determining step in CO oxidation. The lowest energy barrier for the activation of O<sub>2</sub> is occurred at the two adjacent fivecoordinated pits surrounding Ce species (Fig. 15d). Subsequently, the surface-activated oxygen species at the five-coordinated Ce sites can diffuse to the four-coordinated Ce sites and then oxidize the adsorbed CO. Desorption of the generated CO2 is found to recover the catalytic active phase for the oxidation of next CO molecules. In addition, the highly coordination-unsaturated pits surrounding Ce sites would lead to the increase in the hole carrier density and hence assured the rapid CO diffusion along the 2D conducting channel of the surface pits. Importantly, these pit-rich CeO<sub>2</sub> nanosheets delivered a much higher catalytic activity for CO oxidation with a low activation energy of only 61.7 kJ/mol, being much smaller than those of bulk CeO<sub>2</sub> and pit-free CeO<sub>2</sub> nanosheets (Fig. 15h-i).

Besides the recently developed synthetic methods for the optimization of surface properties of CeO<sub>2</sub> [111,112], post-synthesis treatments can also modulate their surface properties in order to promote CO oxidation. Cheung et al. reported the enhanced catalytic activity of nanostructured CeO2 with various morphologies for CO oxidation by using the defect engineering under a low-pressure thermal activation process [88]. Chemical redox etching would induce the morphological evolution of CeO2 nanorods from a smooth to a rough surfaced configuration accompanied with the increased specific surface areas, larger surface Ce<sup>3+</sup> fraction, and higher concentration of oxygen vacancies, thus resulting in the significantly enhanced CO oxidation activity [92]. Therefore, the controlling the surface properties of CeO2 is demonstrated with great potentials to tailor their fine structures, to construct the rationally designed catalytic active sites, and to promote their catalytic activity for CO oxidation.

#### 5.2. CO<sub>2</sub> conversion

At the same time, physical/chemical capture and catalytically converting  $CO_2$  into other chemicals, including useful hydrocarbon fuels and polymers, represent the practical and promising approaches to reduce  $CO_2$  emission as well as to realize its reutilization [113–118]. However, the use of  $CO_2$  as the C1 building block still faces many fundamental and technological challenges because of its chemically inert nature and very high bond energy. In this case, many homogeneous and heterogeneous catalysts have been designed to activate  $CO_2$  at mild conditions. Particularly,  $CeO_2$  catalysts with unique reversible chemical redox oxidation states between  $Ce^{3+}$  and  $Ce^{4+}$  can efficiently trap  $CO_2$  molecules on top of

**Scheme 1.** Transformation of CO<sub>2</sub> to cyclic carbonates, cyclic carbamates, and cyclic ureas from diols, aminoalcohols, and diamines, respectively. Adapted with permission from Ref. [119]. Copyright © 2014, WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim.

the surface basic sites, activate them, and promote the catalytic process in converting  $\mathrm{CO}_2$  into useful chemicals by various techniques, such as high-temperature gaseous phase reactions, liquid phase heterogeneous reactions, and photocatalysis. Generally,  $\mathrm{CeO}_2$  catalysts with the abundant surface defects can potentially enhance their catalytic performance for  $\mathrm{CO}_2$  conversion.

#### 5.2.1. Catalytic synthesis of carbonates by CeO<sub>2</sub> catalysts

Because of the surface bifunctional acid-base properties of  $CeO_2$ , the efficient adsorption of amino groups, hydroxyl groups, and CO<sub>2</sub> would lead to the remarkable catalytic activity and selectivity in production of cyclic carbonates, cyclic carbamates, and cyclic ureas from diols, aminoalcohols, and diamines, respectively, with CO<sub>2</sub> as compared with other transitional metal oxides (ZrO2, CaO, MgO, TiO<sub>2</sub>, etc.) based catalysts [119]. The reaction mechanism catalyzed over CeO<sub>2</sub> is proposed in Scheme 1 [119]. Considering the formation of cyclic ureas as an example, four fundamental steps were involved, as demonstrated in Scheme 2: (1) the co-adsorption of one ethylenediamine molecule and two CO2 molecules at the catalytic active site and the successive formation of the carbamate species on CeO2 surfaces through the surface reaction between ethylenediamine and CO<sub>2</sub>, (2) the transformation of the carbamate species into the surface-adsorbed amine with the release of one CO<sub>2</sub> molecule, (3) the nucleophilic addition of the amine group to the carbamate moiety on CeO<sub>2</sub> and the subsequent formation of the adsorbed 2-imidazolidinone, and (4) the desorption of 2imidazolidinone and the regeneration of CeO2 for further reaction. The third step is usually recognized as the rate-determining step.

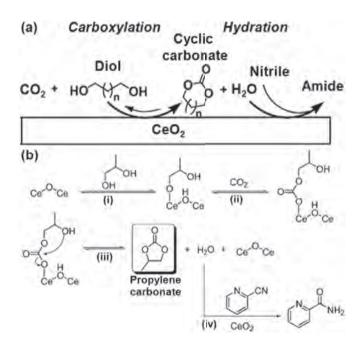
For the synthesis of organic carbonates, especially dimethyl carbonate (DMC), from methanol and  $CO_2$ , the poor conversion efficiency (<10%) and the low chemoselectivity (60–90%) are generally observed even under high temperatures and pressures due to the thermodynamic limitation [120]. Various dehydrating agents are then added to remove the generated water, while a flow apparatus is generally used for the removal of the produced DMC to overcome the abovementioned problems.

For instance, the catalytic activity of CeO<sub>2</sub> for the direct synthesis of DMC in the presence of dehydrating agents was systematically investigated by Tomishige's group [122-124]. A carboxylation/hydration cascade reaction was then realized to synthesize cyclic propylene carbonate, which was catalyzed by CeO<sub>2</sub> with 2-cyanopyridine (Scheme 3a). CeO<sub>2</sub>, as a cascade catalyst, could not only catalytically convert diols into cyclic carbonates but also promoted the removal of water by the hydrolysis of nitrile into amide [121]. In specific, the catalytic mechanism was proposed as the following: (1) the adsorption of diol onto the CeO<sub>2</sub> surface through the interaction between hydroxyl group and Lewis acidic cerium sites, and the subsequent formation of the surface cerium alkoxide species: (2) the insertion of CO<sub>2</sub> into the Ce-O bond to form the surface alkyl carbonate species; (3) the nucleophilic attack of other hydroxyl groups on the carbonyl carbon in carbonate species to generate the products and water, and then consequently recover the surface active sites; and (4) the removal of water by dehydrating chemicals from CeO<sub>2</sub> surfaces (Scheme 3b) [97].

In addition, continuous flowing systems were also developed to further improve the conversion of methanol and the selective transformation of methanol into DMC. In this catalytic system,  $CeO_2$  nanoparticles and 2-cyanopyridine were chosen as the catalyst and the recyclable dehydrating agent, respectively [125]. Controllable  $CO_2$  feeding and stoichiometric amount of methanol and 2-cyanopyridine mixture under various pressures would lead to the high methanol conversion efficiency (>95%) and DMC selectivity (>99%) under the optimized reaction conditions (Fig. 16).

In any case, as efficient heterogeneous catalysts for the direct

Scheme 2. Reaction mechanisms for the synthesis of cyclic carbonate, cyclic carbamate, and cyclic urea over CeO<sub>2</sub>. Adapted with permission from Ref. [119]. Copyright © 2014, WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim.



**Scheme 3.** (a) Carboxylation/hydration cascade catalyst of  $CeO_2$  with nitrile and (b) proposed reaction mechanism. Adapted with permission from Ref. [121]. Copyright  $^{\odot}$  2014, American Chemical Society.

alcohol conversion with  $CO_2$ , the catalytic structure-function relationship has been dedicated with efforts to illustrate the effects of surface properties of  $CeO_2$  on both catalytic activity and selectivity. The yields of DMC were proportional to the specific surface area of catalysts. Stable crystal planes, such as  $CeO_2$  (111), could provide the active catalytic sites [126]. The relationships between the DMC yields and the morphologies of  $CeO_2$  catalysts are then revealed in Fig. 17, which established the consistent correlation among morphology, crystal planes, acid-base sites, and activity of  $CeO_2$ . Also, spindle-like nanosized  $CeO_2$  catalysts, with the large specific surface area, abundant exposed active (111) planes and substantial

amount of acid-base sites, would deliver the highest DMC yield, followed by nanorods, nanocubes, and nanooctahedra with the decreasing yield [127].

The efficient synthesis of DMC could also be achieved by the transesterification method from ethylene carbonate, which was catalyzed by mesoporous CeO<sub>2</sub> catalysts with the high surface area [128]. The surface basic hydroxyl (Ce–O–H) species were performed as catalytic active sites for the adsorption and subsequent activation of CH<sub>3</sub>OH molecules in order to produce the highly active CH<sub>3</sub>O<sup>-</sup> intermediates. CeO<sub>2</sub> has as well been employed in synthesizing other organic carbonates, such as diethyl carbonate (DEC) from CO<sub>2</sub> and ethanol [102,103]. Typically, the amount of surface basic sites and the specific surface areas of CeO<sub>2</sub> were crucial for the adsorption of intermediates. Murzin et al. investigated systematically the effects of surface properties of CeO<sub>2</sub> catalysts synthesized by various methods on their catalytic performances. The DEC yield was observed to be positively related to the amounts of their surface basic sites and their specific surface areas of CeO<sub>2</sub> [129,130].

#### 5.2.2. Reduction of CO<sub>2</sub>

Another effective approach to recycle the heavily discharged CO<sub>2</sub> is to catalytically reduce CO<sub>2</sub> into hydrocarbon fuels (e.g. methane, methanol, and CO) through thermochemical and photocatalytic reactions. In terms of catalysts, as compared with other metal oxides, CeO2 consists of the advantage of oxygen nonstoichiometry, high rates of oxygen chemical diffusivity, facileregulated crystal planes and controllable band gaps, in which all these are demonstrated with potentials for the chemical reduction of CO<sub>2</sub> [131,132]. Moreover, surface control on CeO<sub>2</sub>, by creating more defects, can enhance the visible light absorption, obviously benefiting the corresponding photocatalysis. However, introducing too many defects might lower the photocatalytic efficiency due to the fact that these defects can also acted as the trapping centers of subsequently photogenerated charges. Thus, the optimized surface properties of CeO2 are beneficial for the enhanced catalytic conversion of CO<sub>2</sub>.

Furthermore, the samarium-doped CeO<sub>2</sub> catalysts have also be demonstrated to thermochemically reduced CO<sub>2</sub> with water into hydrogen and CO at high temperatures (over 800 °C) [133]. By using

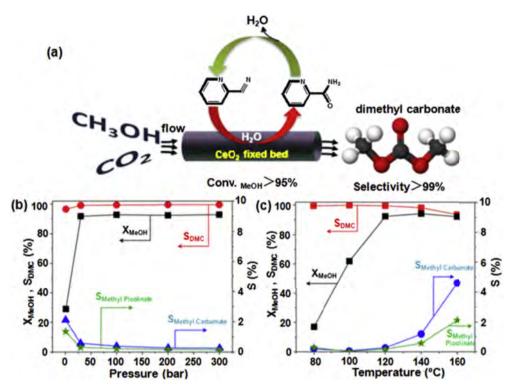


Fig. 16. (a) Scheme of DMC formation using CeO<sub>2</sub> fixed bed and dehydrating agent and effects of (b) pressure and (c) temperature on methanol conversion and product selectivity. Adapted with permission from Ref. [125]. Copyright © 2014, American Chemical Society.

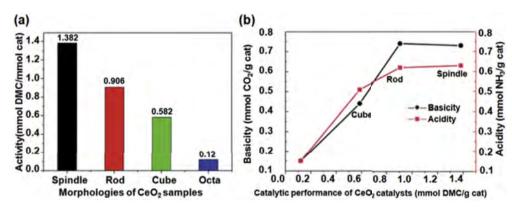


Fig. 17. (a) Catalytic performance of CeO<sub>2</sub> catalysts with different morphologies. (b) Correlation between acid-basicity and catalytic performance of CeO<sub>2</sub> catalysts with different morphologies. Adapted with permission from Ref. [126]. Copyright © 2006, Elsevier B. V. All rights preserved.

this approach, the rapid thermal conversion of CO<sub>2</sub> into useful fuel feedstocks of hydrogen, CO, syngas, or methane could be realized. However, the thermochemical reduction at high temperatures is unfavorable for practical applications because of the poor chemoselectivity, high energy consumption at the endothermic step, and thermal collapse of the catalysts. In this case, the solar-thermal reduction technique was hence considered as the more effective and practical methodology here. Later, a solar cavity-receiver reactor was designed to achieve the high-flux solar-driven thermochemical dissociation of CO<sub>2</sub> and H<sub>2</sub>O with the utilization of nonstoichiometric CeO<sub>2</sub> catalysts [134]. As shown in Fig. 18, a solar reactor was constructed to include a cavity receiver with a windowed aperture in which the concentrated solar radiation could pass through. Porous CeO<sub>2</sub> was then functioned as a catalyst for promoting the dissociation of CO<sub>2</sub> and H<sub>2</sub>O to generate syngas (H<sub>2</sub> and CO) and O<sub>2</sub>, simultaneously.

The thermochemical  $H_2O-CO_2$ -splitting cycle over a non-stoichiometric oxide can be described by the following reactions (Equations (5)-(9)) [134]:

Higher temperature, 
$$T_H$$
:  $1/\delta$   $MO_2 \rightarrow 1/\delta$   $MO_{2-\delta} + 1/2$   $O_2$  (g) (5)

Lower temperature, 
$$T_L$$
:  $H_2O(g) + 1/\delta MO_{2-\delta} \rightarrow 1/\delta MO_2 + H_2(g)(6)$ 

Lower temperature, 
$$T_L$$
:  $CO_2(g) + 1/\delta MO_{2-\delta} \rightarrow 1/\delta MO_2 + CO(g)(7)$ 

Net 
$$H_2O$$
 dissociation:  $H_2O(g) \rightarrow 1/2 O_2(g) + H_2(g)$  (8)

Net 
$$CO_2$$
 dissociation:  $CO_2(g) \rightarrow 1/2 O_2(g) + CO(g)$  (9)

where M represents Ce or the combination of Ce and a dopant element. When the local temperature was over  $900 \,^{\circ}$ C,  $CO_2$ 

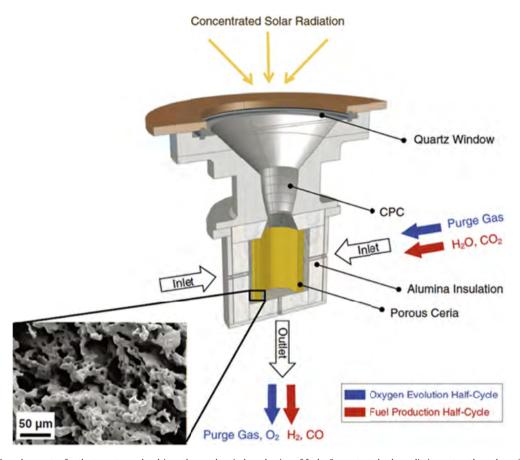


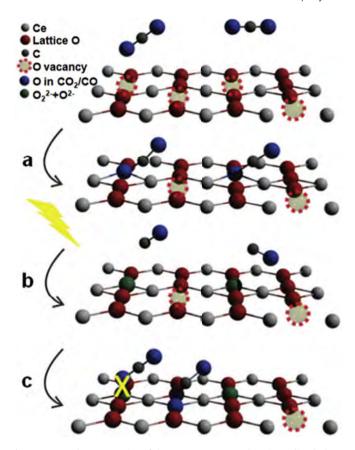
Fig. 18. Schematic of the solar reactor for the two-step, solar-driven thermochemical production of fuels. Concentrated solar radiation enters through a windowed aperture and impinges on the inner walls of ceria. Reacting gases flow radially across the porous ceria toward the cavity inside, whereas product gases exit the cavity through an axial outlet port at the bottom. Adapted with permission from Ref. [134]. Copyright © 2010 American Association for the Advancement of Science.

dissociation was initiated. Then, a selectivity of 100% towards CO production without any detectable amount of carbonaceous species was obtained. Also, the high catalytic stability of porous ceria catalysts could be maintained even after 500 thermochemical cycles such that these catalysts were suitable for practical applications. Typically, the capability of the solar-driven system with a fluidized bed reactor for  $CO_2/H_2O$  conversions was analyzed based on the second law of thermodynamics and the catalytic mechanism of  $CeO_2$  [135]. With the assistance of  $CeO_2$  catalysts and the recapture of waste heat, an increase in the temperature of the carrier gas and the concentration ratio of  $CO_2/H_2O$  would further promote the conversion efficiency of solar to chemical energy up to 43.2%.

In addition, the photocatalytic conversion of CO2 into useful hydrocarbons under mild conditions is also a promising approach [136–142]. In order to suppress the energy barrier for producing CO, the formation of bent  $CO_2^{\delta_{\bullet}}$  intermediates on a semiconductor surface through the efficient activation of CO2 is considered the most important step, followed by the reductive dissociation of the C-O bond. Similar to that in the synthesis of DMC, the catalytic surface properties, including oxygen vacancies and acidity-basicity of CeO<sub>2</sub>, are of great importance in the CO<sub>2</sub> adsorption, diffusion, and electron-facilitated redox chemistry over the catalysts [137]. The defect-enriched CeO<sub>2</sub> nanorods showed the excellent capability of photochemical conversion of CO2 to CO with a high selectivity under ambient conditions [137]. Experimental results also demonstrated that the elimination of surface oxygen vacancies and the enhanced electrophilic species would weaken the CO2 activation, suppressing the migration of the energetic electrons and

terminating the photoreduction of  $CO_2$  (Fig. 19). Thus, the surface oxygen vacancies and the defect-induced local strain accommodated in the defective nanorods were expected to facilitate the activation of  $CO_2$ , to lower the energy barrier, and to provide abundant catalytic sites for the reduction of  $CO_2$ . However, the high concentration of surface defects might inevitably reduce the diffusion of photogenerated charges because of the charge trapping nature of the surface defects.

The photocatalytic activity of CeO<sub>2</sub> for CO<sub>2</sub> reduction is also highly depended on the crystal orientation of the catalysts. For example, a CeO<sub>2</sub> homojunction, consisting of the hexahedron prism-anchored octahedra with the exposed prism surfaces of {100} facets and the exposed octahedral surfaces of {111} facets, could be constructed by the solution-based crystallographic-oriented epitaxial growth (Fig. 20a-d) [140]. Strong structure-activity relationships for the photocatalytic CO<sub>2</sub> conversion were then observed, where Pt was used as the cocatalyst (Fig. 20e-f). This homojunctioned structure were known to contribute to the following three crucial factors: (1) the efficient transfer of photogenerated electrons and holes to the octahedral and prism surfaces, respectively, where this efficient spatial charge separation could promote the separation and duration of photogenerated charges; (2) the CeO<sub>2</sub> hexahedron prism arms as the fast pathways for photogenerated carrier transportation within the homojunction; (3) the different effective mass of electrons and holes on {100} and {111} facets equipped with the higher charge carrier mobility and more facilitated charge separation [113]. As a result, by the rational design of CeO<sub>2</sub> arms, the optimized properties towards methane



**Fig. 19.** Proposed representation of the oxygen vacancy and strain-mediated photochemical CO<sub>2</sub> reduction over rod-like ceria. Reduced ceria (110) is taken for example. Adapted with permission from Ref. [137]. Copyright © 2014, Royal Society of Chemistry.

generation could then be obtained (Fig. 20).

However, the corresponding light adsorption is only limited to the range of ultraviolet and blue region with the wavelength smaller than 360 nm due to the relatively large band gap of CeO<sub>2</sub> (~3.2 eV). In this case, CeO<sub>2</sub> is usually hindered from photocatalysis because of its inappropriate band gap, in which it can only be functioned as the co-catalyst/additive to promote the electron-hole separation and oxygen mobility due to its recyclable Ce<sup>3+</sup>/Ce<sup>4+</sup> pair and OSC [5,90]. As discussed, increasing the surface defect concentration of CeO<sub>2</sub> by controlling the morphology, doping foreign ions, and different post-treatments can apparently narrow down its band gap to broaden the light adsorption as well as to generate more active sites for CO<sub>2</sub> adsorption. Future investigations on the control of surface properties of ceria can further enhance its catalytic activity towards photochemical reduction of CO<sub>2</sub>.

#### 5.3. Organic transformation

As a multi-functional oxide materials, CeO<sub>2</sub> has attracted considerable attentions in the organic catalytic reactions, such as hydrolysis, dehydration, reduction, oxidation, addition, substitution, ring opening, and coupling reactions [6,143–145]. Both the surface-versatile acid-base features and the reversible redox properties of CeO<sub>2</sub> trigger its highly catalytic performances for numerous organic transformations. In general, the surface Ce<sup>3+</sup> and oxygen vacancy have been reported as the active catalytic sites for hydrolysis and oxidation reactions [6]. The surface Lewis basic sites and oxygen vacancy can mediate the electron transfer and modulate the electronic density of the surface active centers, which can

further affect their catalytic capabilities for the reduction and coupling reactions. A wide range of organic reactions catalyzed by CeO<sub>2</sub> have been well summarized by Vivier et al. in 2010 and references therein [6]. Herein, we discuss the recent progress on the catalytic performance of CeO<sub>2</sub> for various organic reactions with the special emphasis on correlations between the surface properties and the catalytic performance.

#### 5.3.1. Hydrolysis and dehydration reactions

Hydrolysis and dehydration reactions, which are of great importance for organic synthesis and industrial production, can be generally catalyzed by various homogeneous Lewis acid catalysts. It is well known that the controllability on the amount and strength of the surface Lewis acid/base is important for these two reactions. For heterogeneous Lewis acid catalysts, the Lewis acid sites will be easily malfunctioned by the chemical dissociation of water on the surface of heterogeneous catalysts during hydrolysis. This irreversible process would lead to the rapid decline of the catalysts' activity. Recent studies demonstrated that CeO2 synthesized by a specific method was belonged to a type of catalyst that is water tolerant during hydrolysis [146]. In this work, CeO2 catalysts showed the exceptionally high activity with a turnover number of up to 260 and a good recyclability for the hydrolysis of 4-methyl-1,3-dioxane to 1,3-butanediol (Fig. 21a and b). The general interaction between water and CeO<sub>2</sub> forms the proton- and hydroxidecovered surface of CeO<sub>2</sub> under hydrothermal conditions or at high temperatures. This irreversible process completely changes the surface acid-base properties of the catalysts. Importantly, the analysis of temperature-programmed desorption of water on the specific CeO<sub>2</sub> catalysts indicated that the water molecules were associatively adsorbed as weak protons and hydroxides on the catalyst surface (type 3 in the profile, Fig. 21c), in which this reversible process was confirmed by the FTIR and nuclear magnetic resonance (NMR) measurements on the acidity of freshly prepared catalysts and water-treated ones. As a result, the CeO<sub>2</sub> catalysts reported in this study are illustrated as water resistant.

The catalytic mechanism for hydrolysis by CeO<sub>2</sub> is also proposed in Fig. 21d. In specific, the associatively adsorbed water located on top of the Lewis acidic sites of CeO2 catalysts can protonate 4methyl-1,3-dioxane, leading to the formation of surface adsorbed acetal and leaving the active site for the associate adsorption of the second water molecule. The second water molecule is then activated at the surface Lewis acidic site of CeO2 catalysts and subsequently hydrolyzes the surface acetal into 1,3-butanediol as the product. Simultaneously, the surface active phase, the Lewis acidic site, is available again for the next hydrolysis process. It is noted that the (111) surface of CeO<sub>2</sub> was confirmed to be the catalytically active crystalline facet for hydrolysis. The surface propertydependent catalytic activity of CeO<sub>2</sub> nanorods was also evaluated for the hydrolysis of 4-methyl-1,3-dioxane, in which the catalysts with the highest concentration of surface defects would yield the highest catalytic activity.

At the same time, hydration of nitriles to amides is another important hydrolysis reaction. In general, this transformation is catalyzed by strong acids or bases. However, these catalysts always induce the over-hydrolysis of amides into carboxylic acids and the formation of salts after the neutralization of catalysts [147]. Similarly, the metal-based homogeneous and heterogeneous catalysts also suffer from restricted reaction conditions, high cost, and difficulties in the separation of product from the reaction systems. In contrast, the surface-engineered CeO<sub>2</sub> could deliver a catalytically selective hydration of 2-cyanopyridine, thereby being capable to produce the corresponding amide at low temperatures (30–100 °C). The surface low-coordinated Ce species or oxygen vacancies of CeO<sub>2</sub> are recognized as the catalytic active phases for

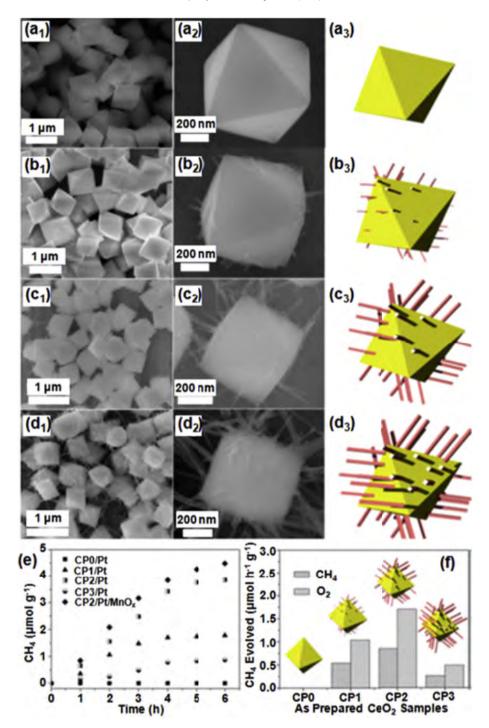


Fig. 20. (a–d) SEM images of homojunction structure of CeO<sub>2</sub> hexahedron prism-anchored octahedra with various hexahedron structures, (e) corresponding specific methane evolution activities, and (f) methane and oxygen evolution rates. Adapted with permission from Ref. [140]. Copyright © 2015, American Chemical Society.

the reaction [148]. As shown in the proposed catalytic mechanism (Fig. 22a), the hydrolysis reaction starts with the dissociation of water molecule on the surface oxygen vacancy sites of  $CeO_2$  to produce  $H^{\delta+}$  and  $OH^{\delta-}$ . Nitrile is then adsorbed on the surface of  $CeO_2$  through the weak interaction between N atom in the ring and surface Lewis acidic cerium sites, forming a nitrile- $CeO_2$  complex. Subsequently, the adsorbed complex undergoes an addition of  $OH^{\delta-}$  to become a nitrile carbon atom, yielding the final amide. This step is typically considered as the rate-determining step. The final desorption of the products always accompanies with the

regeneration of the surface active sites. Based on the catalytic mechanism, one would expect that the post-treatment on  $CeO_2$  catalysts creating more surface defects can further promote the corresponding catalytic activity for hydrolysis of nitriles. When nitriles have a heteroatom (N or O) adjacent to the  $\alpha$ -C of CN group, the rate of hydrolysis reaction is considerably higher than that of other common nitriles (Fig. 22b) [149].

Notably, the nanocrystalline  $CeO_2$  materials have also been found to function as the enhanced catalyst for the dehydration of aldoximes into nitriles as compared with other acid-base metal

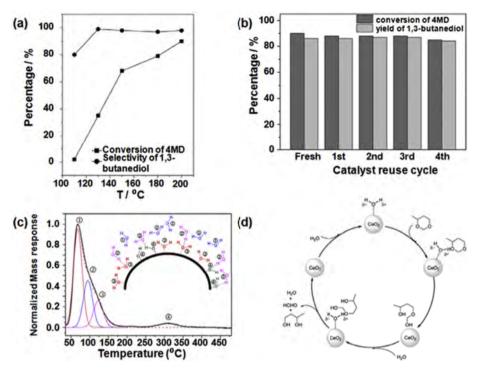


Fig. 21. (a, b) Catalytic performance of CeO<sub>2</sub> for the hydrolysis of 4-methyl-1,3-dioxane to 1,3-butanediol. (c) Profile of temperature-programmed desorption of water from ceria in flowing Ar gas (30 mL min<sup>-1</sup>). The numbers in circles indicate different adsorbed water species on ceria. The inset shows structures of the four types of surface water. (d) Tentative reaction mechanism for the hydrolysis of 4-methyl-1,3-dioxane to 1,3-butanediol. Adapted with permission from Ref. [146]. Copyright © 2013, American Chemical Society.

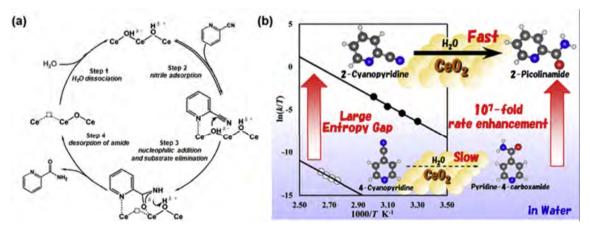


Fig. 22. (a) A proposed mechanism for nitrile hydration on CeO<sub>2</sub>. Adapted with permission from Ref. [148] Copyright © 2013, Royal Society of Chemistry. (b) Catalytic activity of CeO<sub>2</sub> for the hydration of different nitriles. Adapted with permission from Ref. [149]. Copyright © 2015, American Chemical Society.

oxides (e.g. Al<sub>2</sub>O<sub>3</sub>, TiO<sub>2</sub>, and MgO) [150]. Many investigations of their catalytic mechanisms suggest that the Lewis acid sites are generally involved in the adsorption of oxime on the surface of CeO<sub>2</sub> through the interaction between N atoms and surface Lewis acidic sites. The surface Lewis base sites are responsible for catalyzing the C–H bond cleavage [150]. Combining these two results together, the surface acidic and basic sites can be employed for the dehydration of oximes. As discussed above, the nanocrystalline CeO<sub>2</sub> can also be utilized to catalyze the hydrolysis of nitriles. Thus, the picolinamide and picolinic acid alkyl ester derivatives can be obtained from 2-pyridinaldoxime with good yields in the one-pot process.

#### 5.3.2. Oxidation reactions

For oxidation reactions, CeO<sub>2</sub> is as well commonly considered as one of the most promising catalysts because of its superior redox

ability and large OSC. In general, the redox ability of CeO2 is important for the reaction performed at temperatures over 200 °C [151]. Until recently, numerous studies have proved that oxidations catalyzed by CeO<sub>2</sub> catalysts can be achieved at low temperatures (even less than 100 °C) by controlling their morphology, crystal facets, or chemical doping by foreign elements. For instance, CeO<sub>2</sub> alone is observed to exhibit the impressive catalytic activity to oxidize benzyl alcohol into benzaldehyde at 60 °C in presence of air as the oxidant, and the produced benzaldehyde can further couple with aniline to yield imine [152]. The catalytic activity of CeO<sub>2</sub> catalysts for oxidation can be additionally enhanced by loading Au nanocatalysts or Au nanoclusters onto their surface [153]. Au/CeO<sub>2</sub> are demonstrated to catalyze the aerobic oxidation of aromatic anilines to aromatic azo compounds under 100 °C. As compared with the pure CeO<sub>2</sub> catalyst, the catalytic activity of Au/CeO<sub>2</sub> enhances by almost 10 times under the same reaction conditions.

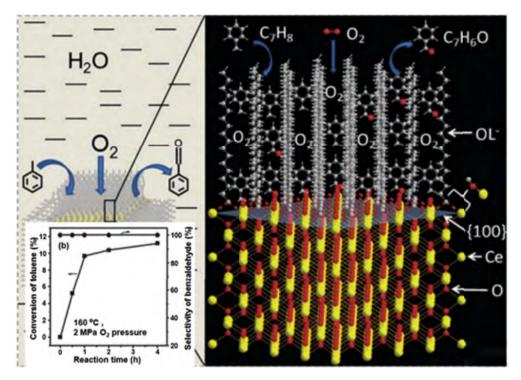
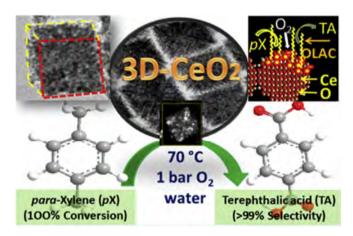


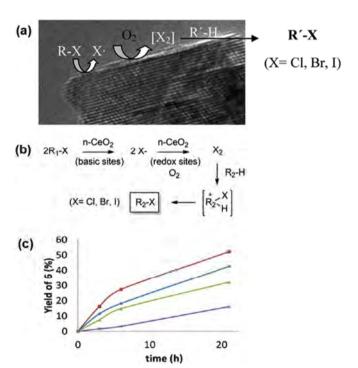
Fig. 23. Oxidation of toluene into benzaldehyde with well-defined CeO<sub>2</sub> nanocubes covered by oleic acid. Adapted with permission from Ref. [158]. Copyright © 2010, Royal Society of Chemistry.



**Fig. 24.** Catalytic oxidation of *para*-xylene into terephthalic acid with 3D self-assembly of  $CeO_2$  nanocubes. Adapted with permission from Ref. [159]. Copyright  $^{\circ}$  2014, American Chemical Society.

Regarding the selective oxidation of benzyl alcohol to benzaldehyde, visible-light-driven oxidations can be occurred even at room temperature catalyzed by Au/CeO<sub>2</sub> catalysts with various morphologies [154–156]. The Mn-doped CeO<sub>2</sub> catalysts with multiple redox states and very high OSC exhibit the superior catalytic performance for the selective oxidation of hydrocarbons under reaction temperatures between 100 and 120 °C [157]. The pronounced catalytic activity of MnCeO<sub>x</sub> catalysts here can be attributed to the formation of Mn<sub>0.5</sub>Ce<sub>0.5</sub>O<sub>x</sub> solid solution, which provides the maximum number of active surface oxygen species for the activation of C–H bonds.

After that, the oxidation of C-H bonds catalyzed by pure  $CeO_2$  will be discussed thoroughly in the following. This reaction imposes a profound challenge for the development of efficient



**Fig. 25.** (a) Oxyhalogenation of activated arenes with nanocrystalline ceria. (b) Possible mechanism for the nanocrystalline ceria-mediated aerobic oxidative halogenation of activated arenes with organic halides. (c) Time-yield plot for the halogenation of 1,3,5-trimethoxybenzene (0.62 mmol) in 1-bromo-3-chloropropane, under oxygen (6 bar, 0.9 mmol) at 140 °C in the presence of 1.2 mmol of ceria nanorods ( $\square$ ), standard nanocrystalline ceria ( $\lozenge$ ), nanoctahedra ( $\triangle$ ), or nanocubes ( $\times$ ). Adapted with permission from Ref. [160]. Copyright  $^{\circ}$  2013, American Chemical Society.

heterogeneous catalysts because of the large bond energy of C–H bonds involved. The well-defined CeO<sub>2</sub> nanocubes covered by oleic

acid are demonstrated with the exclusive selectivity for the oxidation of toluene into benzaldehyde [158]. The surface oleic acids do not only stabilize the  $\{100\}$  crystal facets but also favor the accessibility of organic reactants to the surface of  $CeO_2$  nanocubes in the water media. Importantly, the highly regular surface of  $CeO_2$  nanocubes is characterized as a smooth crystal plane with uniform surface states. Thus, the catalysts can produce only one type of active oxygen, which leads to the exclusively selective oxidation of toluene into benzaldehyde (Fig. 23) [158].

The 3D self-assembly of CeO<sub>2</sub> nanocubes can also be facilitated by a strong coordinating property of oleic acid towards (100) planes of the nanocrystals. The oriented attachment of CeO<sub>2</sub> nanocrystal building blocks forms the (100) surfaces of the exposed porous  $CeO_2$  cubic by sharing the {111} facets (Fig. 24). The assembled  $CeO_2$  nanocubes then exhibited the superior oxidizability for the efficient alkylarene oxidation in water at low temperatures (below 85 °C) and 1 bar  $O_2$  pressure [159]. The features of high concentration of surface oxygen vacancies, small crystalline size of cubic building blocks, large surface area and pore size, and large quantity of hydrophobic oleic acid surfactants are believed to play important roles in the catalytic oxidation of alkylarenes. In this structure, the oxygen vacancy is the catalytically active site for the oxidation. As compared with  $CeO_2$  (111), the (100) facets favor the formation of oxygen vacancy and deliver a large value of OSC. The long tail of

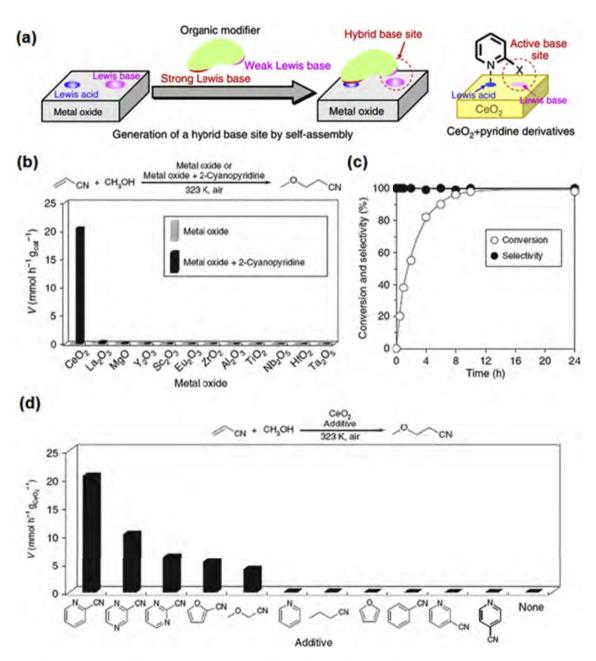


Fig. 26. Metal oxide and nitrile screening and the time course. (a) Comparison of the combination of metal oxide catalysts and 2-cyanopyridine. Reaction conditions without 2-cyanopyridine (gray bar): acrylonitrile (10 mmol), methanol (20 mmol), metal oxide (172 mg), 323 K, air, 12–48 h. Reaction conditions with 2-cyanopyridine (black bar): acrylonitrile (10 mmol), methanol (20 mmol), metal oxide (172 mg), 2-cyanopyridine (2 mmol), 323 K, air, 0.5–48 h. (b) Time course of the reaction over CeO<sub>2</sub>+2-cyanopyridine hybrid catalyst. Reaction conditions: acrylonitrile (10 mmol), methanol (15 mmol), CeO<sub>2</sub> (1 mmol), 2-cyanopyridine (1 mmol), 323 K, air. (c), Effect of organic compounds on the catalytic activity. Reaction conditions: acrylonitrile (10 mmol), methanol (20 mmol), CeO<sub>2</sub> (1 mmol), organic additive (2 mmol), 323 K, air. Adapted with permission from Ref. [161]. Copyright © 2015, Rights Managed by Nature Publishing Group.

hydrophobic oleic acid can also trap the alkylarenes into the pores and help them to reach the catalyst surface. The defective sites on the catalyst surface can as well promote the formation of intermediates, which further react with oxygen to release the products. After the removal of the surfactants, the catalysts are found to give the very poor catalytic activity and selectivity.

In any case, CeO<sub>2</sub> nanostructures are also revealed to deliver the remarkable catalytic activity for the oxidative oxyhalogenation of arenes under aerobic conditions [160]. Halogenation of hydrocarbons is an important process in the chemical industry. As illustrated in Fig. 25a, the C-H bond of the arenes can be activated effectively by CeO<sub>2</sub> catalysts and halogenated by the organic or inorganic halogen compounds in the presence of oxygen. The possible reaction mechanism is shown in Fig. 25b. Initially, the surface basic species of CeO<sub>2</sub> activates the halogen compounds to produce anionic halides, where the halogen radicals can be later generated at the surface redox sites of CeO<sub>2</sub>. When the two radicals interact with the dihalogen species, the products can be obtained by an electrophilic aromatic substitution. Notably, the formation of CeO<sub>2</sub>halogen intermediates is also possible. Thus, regulating the surface properties of CeO<sub>2</sub> catalysts is also important for the enhancement of their catalytic activities. CeO<sub>2</sub> nanoparticles with the smaller crystalline size can as well give the higher catalytic activity, suggesting that the available surface areas and the abundance of surface defects contribute dominantly to the oxidative reactions. The plane orientations also play important roles for the catalytic oxidative halogenation of arenes. Among all the different morphologies, CeO<sub>2</sub> nanorods gave the best catalytic performance for oxidation reactions.

#### 5.3.3. Hydromethoxylation of acrylonitrile

As compared with the unmodified catalysts, creating new catalytic active sites on the surface of heterogeneous catalysts by using small molecules as modifiers can improve their catalytic efficiency and chemoselectivity dramatically [161]. For example, there is a recent work employing pyridine and its derivatives as organic modifiers to construct a novel charge-transfer complex as the hybrid interface basic site on the surface of CeO<sub>2</sub> nanoparticles (Fig. 26a). In this strategy, the surface Lewis acidic sites of CeO<sub>2</sub> provide the adsorption sites for the organic modifiers. The defectabundant surface of CeO2 exhibits a structural feature with the adjacent surface Lewis acidic and basic sites. When the organic modifiers are existed in the presence of Lewis basic functional groups, the surface modification would produce a new Lewis base configuration composed of two bases, with one from the organic modifiers while another one from the surface adjacent Lewis bases. Once 2-cyanopyridine is used as the organic modifier, such a structure can deliver a 2000-fold enhancement for the hydromethoxylation of acrylonitrile with a very high selectivity as compared with CeO<sub>2</sub> alone (Fig. 26b). This impressive catalytic behavior is only observed on the complex of CeO<sub>2</sub> and pyridine modifiers, suggesting the importance of unique acid-base properties of CeO<sub>2</sub> for the formation of new surface catalytic active sites. Among various organic modifiers, 2-cyanopyridine is considered to be the most effective one (Fig. 26c).

#### 5.3.4. Catalytic hydrogenation

In general, CeO<sub>2</sub> can serves as the promoter, stabilizer, or additive to improve the hydrogenation catalytic activity of anchored metal active phases (e.g. Pt, and Pd) [2]. The strong metal-support interaction effects between CeO<sub>2</sub> and metal active phases can enrich the electron density of supporting metals and subsequently promote their capability for hydrogen activation and dissociation [2,162,163]. Combining with the acid-base and redox properties of CeO<sub>2</sub>, the catalytic activity and selectivity of metal/CeO<sub>2</sub> catalysts

for hydrogenation reactions can be dramatically modulated [162,164]. With CeO<sub>2</sub> as the support, Cu and Au catalysts also exhibit the strong ability for hydrogen activation and deliver the excellent catalytic activity for hydrogenation of unsaturated compounds [165,166]. In particular, Cu/CeO<sub>2</sub> catalysts are revealed with the good activity towards the hydrogenation of DMCs to methanol at a temperature of 160 °C and a low H<sub>2</sub> pressure of 2.5 MPa [167]. Because of the large amount of surface defects, specific organic groups will be preferentially adsorbed on the surface of CeO<sub>2</sub> rather than the surface of metal nanocatalysts. This highly selective adsorption model of substrates may also dramatically promote the catalytic hydrogenation performance of CeO<sub>2</sub>-based catalysts. Recently, the subnano Pd clusters anchored on the surface defectabundant PN-CeO<sub>2</sub> were shown to deliver the enhanced catalytic efficiency and chemoselectivity for the hydrogenation of nitroarenes [163]. For the hydrogenation of 4-nitrophenol, the catalyst yielded a TOF of ~44059 h<sup>-1</sup> and a chemoselectivity to 4aminophenol of >99.9%. This superior catalytic performance could be attributed to a cooperative effect between the highly dispersed subnano Pd clusters for the hydrogen activation and the unique surface sites of PN-CeO<sub>2</sub> with a high concentration of oxygen vacancy for the energetically and geometrically preferential adsorption of nitroarenes via the nitro group.

In fact, the catalytic activity of pure  $CeO_2$  is commonly observed for oxidations rather than reductive hydrogenations. Previous DFT calculations indicated that  $CeO_2$  (111) can effectively activate hydrogen molecules with a low activation energy of 0.2 eV and a high exothermicity ( $-2.82\,eV$ ) [168]. The adsorption energies of hydrogen species on  $CeO_2$  (110) and  $CeO_2$  (111) surfaces are calculated to be -150.8 and  $-128.3\,kJ/mol$ , respectively. These simulation results indicated that the  $H_2$  molecule can be effectively activated and energetically adsorbed on the  $CeO_2$  surface. Early in 2007, it was reported that pure  $CeO_2$  nanoparticles could hydrogenate benzoic acid into benzaldehyde at  $380\,^{\circ}C$  in a fixed bed reactor [169]. Their rapid deactivated performance was attributed to the coke formation on the surface of  $CeO_2$  nanoparticles; however, there is not any detailed catalytic mechanism explored for benzoic acid hydrogenations catalyzed by  $CeO_2$  [169,170].

Lately, pure CeO<sub>2</sub> nanoparticles were discovered to exhibit their high catalytic activity and chemoselectivity in the partial hydrogenation of alkynes into olefins [171]. The hydrogenation of propyne at ambient pressure in the gas phase was used as a demonstration [141]. The hydrogenation conditions were optimized at 523 K with a H<sub>2</sub>/C<sub>3</sub>H<sub>4</sub> ratio of 30:1 and a contact time of 0.21 s. The conversion of propyne could reach a maximum of 96% with a selectivity to propene of 91%. An increase in the partial pressure of H<sub>2</sub> was positively related to both hydrogenation activity and chemoselectivity. However, either an increase or decrease in the reaction temperature would lead to a decreased catalytic activity. Also, the chemoselectivity of propene is generally decreased with the increasing reaction temperature. It is recognized that the mutative surface properties of CeO<sub>2</sub> catalysts under the operation conditions were critical to understand their hydrogenation performance. The propyne conversion was strongly dependent on the calcination temperature of catalysts (Fig. 27a), while the chemoselectivity to propene was moderately influenced by the calcination temperature. This catalytic phenomenon can be attributed to the decrease of the surface area of CeO<sub>2</sub> catalysts as a result of the increase in the calcination temperature. When the catalysts were pretreated with H<sub>2</sub>/He, both the hydrogenation activity and selectivity decreased dramatically (Fig. 27b), which suggested that the surface reduction levels of CeO<sub>2</sub> catalysts were important to their catalytic performance.

At the same time, the corresponding catalytic mechanism can be proposed from the results of operando infrared spectroscopy. In the

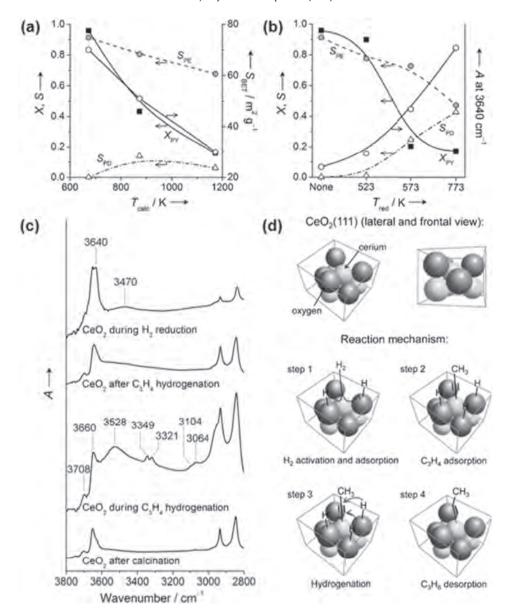
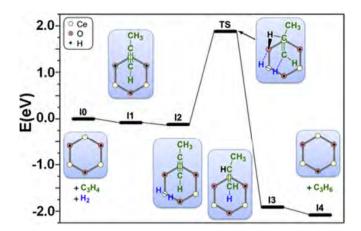


Fig. 27. (a) Conversion of propyne (XPY) and selectivity to propene (SPE) and propadiene (SPD) at 1 bar as a function of (a) the calcination temperature and (b) the reduction temperature of CeO<sub>2</sub>. Reaction conditions:  $H_2/C_3H_4$  ratio of 30:1, T=523 K, and t=0.21 s. The influence of the calcination temperature on a specific surface area of the solid ( $S_{BET}$  determined by  $N_2$  adsorption) and the reduction temperature on the extent of surface vacancies (determined by infrared spectroscopy; A=absorbance) are plotted in the secondary  $\gamma$  axis of (a) and (b), respectively. (c) DRIFT spectra of CeO<sub>2</sub> recorded at 523 K, after calcination of propyne (in He flow), during hydrogenation of propyne ( $H_2/C_3H_4$  ratio of 30:1 and the proposed reaction mechanism. Adapted with permission from Ref. [171]. Copyright  $\Omega$  2012, WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim.

first step, hydrogen molecules are activated on the surface oxygen of CeO<sub>2</sub>, whereas two surface OH groups are formed as a consequence (Fig. 27d). The strong surface acidic and basic sites of CeO<sub>2</sub> can then be employed to dissociate propyne into methylacetylide (CH<sub>3</sub>–C≡C) on top of the cerium and another OH group. The evolution of diffuse reflection infrared Fourier transform (DRIFT) spectra provides evidence of the proposed activation process (Fig. 27c). The activated hydrogen species hydrogenates the methylacetylide into propene (step 3). Then, the desorption of propene recovers the surface properties of CeO<sub>2</sub> catalysts (step 4). As a result, the surface area and oxidation states of CeO<sub>2</sub> catalysts determine their catalytic activity and chemoselectivity for hydrogenation of alkynes into alkenes. The high reduction degree of CeO<sub>2</sub> catalysts would generate a high concentration of oxygen vacancy, which minimizes the available surface sites for the dissociation of

alkynes, leading to a decrease in the catalytic activity. The surface oxygen species are also crucial for the stabilization of activated hydrogen species for the next hydrogenation. In this case, CeO<sub>2</sub> catalysts with the exposed facets with low vacancy content can benefit the hydrogenation of alkynes into alkenes. The higher catalytic performance of CeO<sub>2</sub> is then typically dominated by the (111) surface, as compared to that of CeO<sub>2</sub> nanocubes enclosed with (100) facets [172].

However, the exact catalytic mechanism is still under debate till now. A recent DFT calculation study suggested a concerted pathway for the hydrogenation of alkynes (Fig. 28) [173]. The high ratio of  $H_2/C_3H_4$  makes the  $CeO_2$  surface completely hydroxylated. Both  $H_2$  and  $C_3H_4$  can be adsorbed on the top of  $Ce^{3+}$ . In this approach, the adsorbed  $H_2$  molecule, the two carbon atoms from unabsorbed alkynes, and one OH group form a six-member ring structure as the



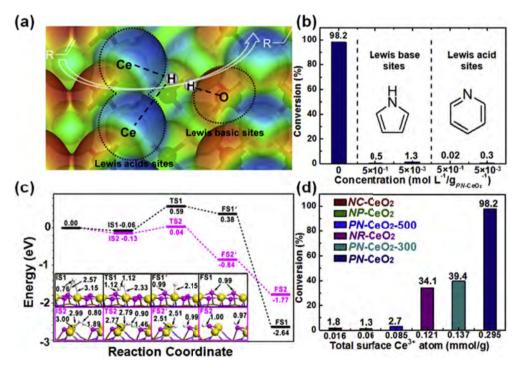
**Fig. 28.** Overall reaction energy profile for the selective hydrogenation of propyne. For the sake of clarity, the fully hydroxylated  $CeO_2(111)$  surface is represented by a gray hexagon, where the Ce, O, and H surface atoms are depicted as pale yellow, red, and black spheres, respectively. The energy of TS has been corrected by the ZPVE. Adapted with permission from Ref. [173]. Copyright  $^{\circ}$  2014, American Chemical Society. (For interpretation of the references to color in this figure legend, the reader is referred to the Web version of this article.)

transition state. The calculated activation energy of 1.88 eV indicated that the proposed concerted pathway is plausible under the calculated conditions. It should also be noted that this hydrogenation mechanism is proposed on the basis of highly H-covered surface of  $\text{CeO}_2$  catalysts.

In general, high temperature is required for the hydrogenation catalyzed by  $\text{CeO}_2$  because of its inherent low capability of the heterolytic (between lattice  $\text{Ce}^{4+}$  and  $\text{O}^{2-}$ ) pathway for  $\text{H}_2$ 

dissociation. Moreover, the substrate selection is limited to alkynes with the terminal C=C triple bond because of the prerequirements of the dissociation of -C≡C-H at the catalyst surface. Recently, PN-CeO<sub>2</sub> catalysts with regulated surface properties have been reported to deliver a very high catalytic activity for the hydrogenation of alkenes and alkynes with the wider scope under mild conditions (Fig. 29a) [174]. As compared to others in the literature, three important factors were outlined: (1) PN-CeO<sub>2</sub> catalysts have a very high concentration of the surface defects, (2) both alkynes and alkenes can be reduced into alkanes, (3) the reaction conditions ( $T = 100 \,^{\circ}\text{C}$  and  $P(H_2) = 1.0 \,\text{MPa}$ ) are much milder than those specified in the literature. Moreover, DFT calculations and control experiments suggest that the richness of surface defects is critical for the construction of a new surface Lewis acidic center by two adjacent reduced surface Ce atoms near the oxygen vacancy. The "fixed" surface lattice oxygen as Lewis base and the constructed Lewis acid have a large possibility to be close enough but independent due to the richness of surface defects as well as the unique geometrical and electronic configurations, being analogy to the molecular homogeneous FLPs. The theoretical results as well demonstrated that the created FLP sites can be easily used to dissociate H-H bond with a low activation energy of 0.17 eV (Fig. 29c).

The necessity of the co-existence of constructed Lewis acidic and basic sites of *PN*-CeO<sub>2</sub> for the hydrogenation of styrene was further confirmed in the presence of other molecular Lewis acid or base during the catalytic process. The addition of very small amount of Lewis base pyridine and Lewis acid pyrrole could completely terminate the hydrogenation activity of *PN*-CeO<sub>2</sub> (Fig. 29b). The blockage of surface sites of *PN*-CeO<sub>2</sub> by trace amount of pyridine/pyrrole molecules were found to give a low possibility



**Fig. 29.** (a) Schematic of created "FLP" catalytic sites on the surface of CeO<sub>2</sub>. (b) Energy profile for  $H_2$  dissociation on ideal CeO<sub>2</sub>(110) shown by the black curve and on CeO<sub>2</sub>(110) with two oxygen vacancies shown by red curves. The optimized structures of initial states (IS), transition states (TS), and final states (FS) are labeled with bond distance (in Å). The zero energy reference corresponds to the sum energy of  $H_2$  in the gas phase and the corresponding clean CeO<sub>2</sub> surfaces. (c) Effects of molecular Lewis base or Lewis acid on the catalytic activity of PN-CeO<sub>2</sub> for the hydrogenation of styrene. (d) Effects of the total number of surface  $Ce^{3+}$  atoms on the hydrogenation of styrene. NC-CeO<sub>2</sub> (Nanocube CeO<sub>2</sub>), PN-CeO<sub>2</sub> (Nanoparticle CeO<sub>2</sub>), PN-CeO<sub>2</sub>-500 (Porous CeO<sub>2</sub> nanorod calculation under air at 500 °C for 10 h), NR-CeO<sub>2</sub> (Nanorod CeO<sub>2</sub>), PN-CeO<sub>2</sub>-300 (Porous CeO<sub>2</sub> nanorods). Adapted with permission from Ref. [174]. Copyright  $^{\circ}$  2017, Rights Managed by Nature Publishing Group. (For interpretation of the references to color in this figure legend, the reader is referred to the Web version of this article.)

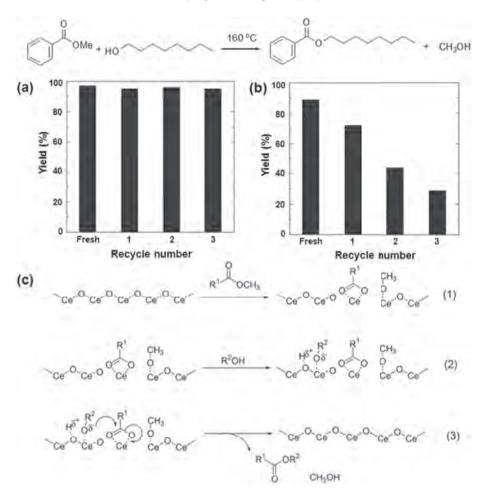


Fig. 30. Recycle tests of (a) CeO<sub>2</sub> and (b) CaO for the transesterification of methyl benzoate with n-octyl alcohol. (c) A proposed mechanism of transesterification by CeO<sub>2</sub> catalysts. Adapted with permission from Ref. [177]. Copyright © 2013, Royal Society of Chemistry.

for the formation of surface FLP sites in the aspect of population of adjacent surface  $Ce^{3+}$ . Fig. 29d shows the conversion efficiency of styrene as a function of the total amount of the surface  $Ce^{3+}$  species per gram of various  $CeO_2$  catalysts, which evidently demonstrates the importance of surface defects for the construction of surface FLP-like active sites and their consequent hydrogenation activity.

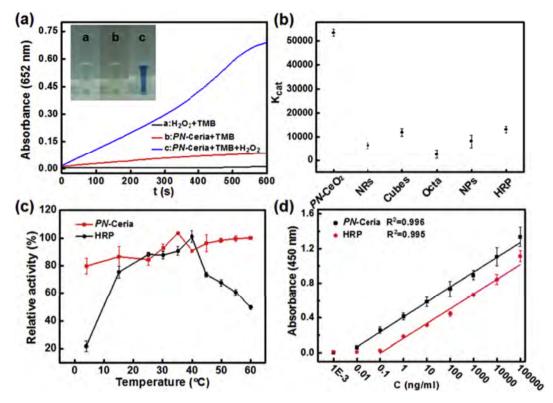
In addition,  $CeO_2$  nanorods covered by {110} planes are also observed to efficiently and selectively catalyze the hydrogenation of nitroaromatics by using  $N_2H_4$  as the reducing agent [175]. These {110} planes usually exhibit the higher catalytic activity for the hydrogenation as compared with the {100} and {111} counterparts. Theoretical studies showed that the surface reduction reaction on the {110} planes is energetically much more favorable because the oxygen vacancies are more easily formed on the {110} planes than on the {111} and {100} planes [175].

All the above hydrogenation performances of CeO<sub>2</sub> suggest that the surface properties of catalysts play very important roles in the reaction. Initially, the surface defects of CeO<sub>2</sub> are believed to be detrimental for the hydrogen activation [171,172]. Surprisingly, further increasing the surface defects of CeO<sub>2</sub> would lead to the formation of defect clusters and the construction of new surface active sites of the solid frustrated Lewis pairs, which can be used efficiently for the hydrogen activation and subsequent hydrogenation [174,176]. Such a strategy is of great interest to develop high-performance heterogeneous catalysts with the modulated active sites by regulating their surface properties.

#### 5.3.5. Other catalytic organic reactions

Apart from conventional organic reactions, pure CeO2 nanostructures have also been explored for many other organic catalysis. The transesterification of esters with alcohols is generally catalyzed by homogeneous protic acids, Lewis acids, and basic catalysts. Although heterogeneous catalysts (e.g., CaO, MgO, and polyoxometalates) have been reported for transesterification reactions, most of them are suffered from the catalyst leaching and narrow substrate scopes. Inspired from the co-existence of both surface acidic and basic species, CeO<sub>2</sub> was lately developed as the robust heterogeneous catalyst for the solvent-free transesterification of esters [177]. As compared with the catalytic performance of CaO, CeO<sub>2</sub> nanoparticles exhibited the higher catalytic efficiency with the good recyclability for transesterification of esters (Fig. 30a and b). Also, a catalytic mechanism of transesterification by CeO2 is proposed in Fig. 30c. In the first step, the ester molecule is cleaved and adsorbed on the top of surface cerium sites, yielding the surface carboxylate species. Assisted by the surface basic sites of CeO<sub>2</sub>, the dissociative adsorption of alcohol is also occurred at the catalyst surface to form the alkoxide species. The subsequent nucleophilic substitution between surface alkoxides and carboxylate species as well as the corresponding desorption would leave the products. It is noted that the proton abstraction of alcohol by a Lewis base site of CeO<sub>2</sub> is considered the rate-limiting step. Thus, the high concentration of surface basic sites will be expected to promote the catalytic performance of CeO<sub>2</sub>.

Furthermore, CeO<sub>2</sub> nanoparticles also exhibited their catalytic



**Fig. 31.** Peroxidase A. Peroxidase-like activity of *PN*-Ceria. (a) Photograph of the mixture of TMB and  $H_2O_2$  in the absence of *PN*-Ceria. (b) Photograph of the mixture of TMB and *PN*-Ceria. (c) Photograph of the mixture of TMB,  $H_2O_2$ , and *PN*-Ceria. Blue color was observed. B. Comparison of the  $K_{cat}$  for ceria nanomaterials and HRP. C. The temperature-dependent peroxidase-like activity of *PN*-Ceria and HRP. The pH and  $H_2O_2$  concentration were 4.0 and 100 mM, respectively. The maximum point in each curve was set as 100%. D. Comparison of the detection limit of *PN*-Ceria and HRP system. Adapted with permission from Ref. [183]. Copyright © 2015, Elsevier B. V. All rights preserved. (For interpretation of the references to color in this figure legend, the reader is referred to the Web version of this article.)

activity for the one-step synthesis of esters from nitriles and alcohols [178]. The initial step of the catalytic reaction is similar to that of the selective hydrolysis of nitriles into amides. The surface amide-CeO<sub>2</sub> complex then undergoes an addition of alcohol to become the amide carbon atom to yield the ester. Likewise, the surface defects determine the catalytic activity. Thus, CeO<sub>2</sub> catalysts could deliver more than two orders of magnitude higher activity than other metal oxides (TiO<sub>2</sub>, SnO<sub>2</sub>, MnO<sub>2</sub>, and ZrO<sub>2</sub>, etc.), which was attributed to the unique surface acid-base and reversible redox properties of CeO<sub>2</sub>. In another case, the Pr-droped CeO<sub>2</sub> catalysts showed the enhanced catalytic activity to convert isobutene, formaldehyde, and water into 3-methyl-1,3-butanediol in one step as contrasted to that enabled by pure CeO<sub>2</sub>. This enhancement is largely due to the fact that the Pr doping can create more surface oxygen vacancies on the catalyst surface and subsequently improve the Lewis acidity of catalysts for the promoted catalytic activity [178].

#### 5.4. Biomimetic catalysis

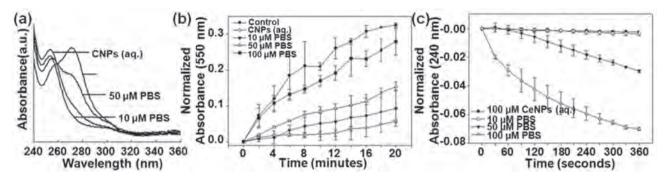
In addition,  $CeO_2$  nanostructures have as well been extensively explored to mimic various natural enzymes, such as superoxide dismutate (SOD), peroxidase, catalase, and oxidase [179–184]. The multiple enzyme-like properties of  $CeO_2$  can contribute diverse applications for clinic diagnosis, biosensing, drug delivery, and antioxidants against oxidative stress [185–198]. The use of  $CeO_2$  nanostructures as artificial nanozymes has been originated from their surface reversible mutation of the oxidation states between  $Ce^{3+}$  and  $Ce^{4+}$  [179,181,182,199] as well as their features of biocompatibility and nontoxic nature [200,201]. Therefore, the

surface defects of CeO<sub>2</sub> typically serve as the catalytic active sites. Regulating their surface by chemical techniques can considerably affect the catalytic performances of CeO<sub>2</sub> nanostructures as artificial enzymes [179]. Comprehensive reviews on the artificial nanozymes, including CeO<sub>2</sub>, have been performed. In this work, we aim to focus on the recent progress of biomimetic catalytic behavior of nanoceria, especially on those through the surface control.

#### 5.4.1. Peroxidase-like activity of CeO<sub>2</sub>

Peroxidases are a class of natural enzymes that catalyze the oxidation of compounds by the decomposition of hydrogen peroxide or organic peroxide [202]. Despite that  $CeO_2$  nanostructures are widely explored as artificial SODs, oxidases, and catalases as well as for potential applications in biology [184,203,204], their peroxidase activity are seldom investigated because of their relatively low peroxidase activity [205]. Previous catalytic mechanism studies suggested that the surface properties of  $CeO_2$  with a large surface  $Ce^{3+}$  fraction would play an important role to boost the peroxidase activity since the peroxidase-like activity of  $CeO_2$  is originated from their ability of  $Ce^{3+}/Ce^{4+}$  rapid cycling [182,183]. In this case, the highest available amount of  $Ce^{3+}$  on the surface of  $CeO_2$  may promote the reaction efficiency and catalytic activity.

Recently, we reported that *PN*-CeO<sub>2</sub> displayed a very strong peroxidase-like activity by catalyzing the oxidation of 3,3′,5,5′-tetramethylbenzidine (TMB) in the presence of H<sub>2</sub>O<sub>2</sub> (Fig. 31a), which demonstrated the importance of surface properties of CeO<sub>2</sub> as artificial nanozymes on their catalytic performance [183]. As compared with the natural horseradish peroxidase (HRP) and other nanocerias with various morphologies (*e.g.* nanoparticles,



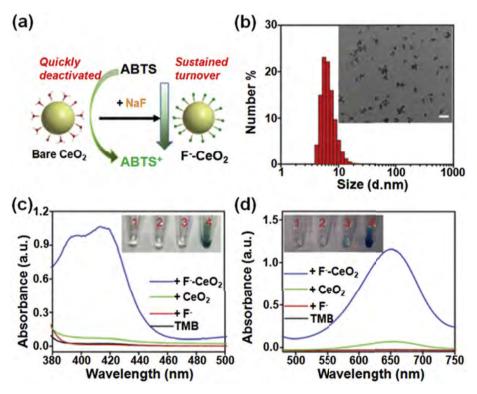
**Fig. 32.** SOD and catalase dose-dependent shift in surface oxidation state of CeNPs by treatment with phosphate anions (PO<sub>4</sub>). CeNPs (100 mM) were dispersed in a range of phosphate concentrations (10, 50, and 100 mM) and incubated for 24 h before UV–visible spectroscopy (a), SOD mimetic (b) and catalase mimetic (c) activity measurements from the resulting suspensions. A kinetic measurement of catalase-like activity was analyzed by following H<sub>2</sub>O<sub>2</sub> degradation at 240 nm. Adapted with permission from Ref. [207]. Copyright © 2011, Elsevier B. V. All rights preserved.

nonporous nanorods, cubes and octahedra), apparent steady-state kinetic studies indicate the best affinity of TMB with PN-CeO<sub>2</sub> and the highest catalytic activity of PN-CeO<sub>2</sub> as the peroxidase (Fig. 31b). Kinetic studies also suggest that the defective sites can serve as catalytic centers with the enhanced ability for activation of H<sub>2</sub>O<sub>2</sub>. Thus, PN-CeO<sub>2</sub> with the highest surface Ce<sup>3+</sup> fraction and the largest oxygen vacancy could deliver the superior peroxidase-like activity. This also explains the very low catalytic activity of commonly reported CeO<sub>2</sub> nanoparticles. Another unique aspect of PN-CeO<sub>2</sub> as the artificial peroxidase is its almost constant activity at various measurement temperatures of 4–60 °C (Fig. 31c). In addition, PN-CeO<sub>2</sub> as the artificial peroxidase shows a very high chemical and catalytic stability under harsh conditions. Later, an accurate and reliable immunoassay based on PN-CeO<sub>2</sub> was also

demonstrated for the specific detection of CA15-3 (a biomarker of breast cancer, Fig. 31d). The detection limit of *PN*-CeO<sub>2</sub> was 0.01 ng/mL, which was one order of magnitude higher than that of the HRP system.

#### 5.4.2. SOD and catalase activity

SOD is a natural enzyme that catalyzes the dismutation of superoxide radicals into either  $O_2$  or  $H_2O_2$ . Catalase is a common enzyme that catalyzes the decomposition of  $H_2O_2$  to  $H_2O$  and  $O_2$ . Thus, both enzymes have protective effects against the oxidative stress induced by strong oxidants and exhibit the potential to protect the living organisms from the oxidative stress [186–188]. When  $CeO_2$  mimics the SOD enzymes, the produced  $H_2O_2$ , as the source of hydroxyl radicals (i.e. the most deconstructive ROS), is



**Fig. 33.** Oxidase mimicking activity of nanoceria by fluoride capping. (a) A scheme showing F<sup>-</sup>-capped nanoceria with improved oxidase turnovers. (b) DLS size distribution and a TEM image (inset) of nanoceria. The scale bar in the inset of (b) is 20 nm. UV–vis spectra of (c) ABTS (0.5 mM) and (d) TMB (1 mM) oxidation by nanoceria (100 μg mL<sup>-1</sup>) with or without F<sup>-</sup> (1 mM for ABTS, 5 mM for TMB) at pH 4 (acetate buffer, 20 mM) after 30 min of reaction. The insets are 1: the free substrate, 2: the substrate with F<sup>-</sup>, 3: the substrate with bare CeO<sub>2</sub>, 4: the substrate with F<sup>-</sup>-capped CeO<sub>2</sub>. Adapted with permission from Ref. [211]. Copyright <sup>©</sup> 2016, Royal Society of Chemistry.

considered as the more toxic species over superoxide for biological systems [179,187,206]. Fortunately, CeO<sub>2</sub> artificial nanozymes can also serve as catalases and decompose  $\rm H_2O_2$  into harmless  $\rm O_2$  and  $\rm H_2O$ . As a result, CeO<sub>2</sub> nanozymes are promising to be used as antioxidants for the therapy involving oxidative stress. Many factors, including the morphology, size, surface  $\rm Ce^{3+}/\rm Ce^{4+}$  ratio, and reaction conditions (*e.g.*, pH, buffer species), affect the SOD and catalase activity of CeO<sub>2</sub> [181,182]. It has been recognized that the high surface  $\rm Ce^{3+}/\rm Ce^{4+}$  ratio of CeO<sub>2</sub> nanozymes is a positive factor for their SOD-like activity but deconstructive for their catalase-like activity [182]. Thus, it is important to control the surface properties of CeO<sub>2</sub> to combine the SOD and catalase mimetic processes together, where the generated  $\rm H_2O_2$  from SOD mimetic system can be immediately consumed by the catalase mimetic system.

At the same time, inorganic ions, such as phosphate ions  $(PO_4^{3-})$ , have been recently found to effectively modulate the SOD and catalase activity of  $CeO_2$  nanoparticles [207]. The phosphate ions are strongly adsorbed on the surface of cerium as a hard Lewis acid. When the concentration of phosphate ions is increased in the reaction environment, the SOD-like activity of  $CeO_2$  is gradually inhibited but the catalase-like activity is enhanced (Fig. 32). Thus, it provides a practical approach to reach the equilibrium between  $H_2O_2$  generation from the SOD process and subsequent  $H_2O_2$ 

decomposition from catalase process.

#### 5.4.3. Oxidase activity

Oxidase is an enzyme that catalyzes the oxidation-reduction reaction involving  $O_2$  as the electron acceptor, in which  $O_2$  is reduced to  $H_2O$  or  $H_2O_2$ . In general, the oxidase-like activity of  $CeO_2$  nanozymes has been evaluated and then further developed to use as colorimetric sensors for the selective detection of dopamine and catechol [208–210]. Similarly, the oxidase mimetic activity of  $CeO_2$  is also considerably affected by the surface properties of artificial enzymes. Fluoride-treated  $CeO_2$  nanoparticles showed over a 100-fold enhancement on their oxidase-like activity, as compared with the untreated counterparts (Fig. 33) [211]. This vast enhanced catalytic activity can be attributed to the alternated surface charge after the fluoride modification, in which the strong electron withdrawing properties of fluoride ions can promote the efficient adsorption of the substrates and desorption of the products.

#### 5.4.4. Phosphatase-mimetic activity

In addition, phosphatase is an enzyme that hydrolyzes the phosphoric acid monoester into a phosphate ion and a free hydroxyl group [212]. Actually, phosphatase enzymes are essential to many biological functions, especially in the cellular regulation and

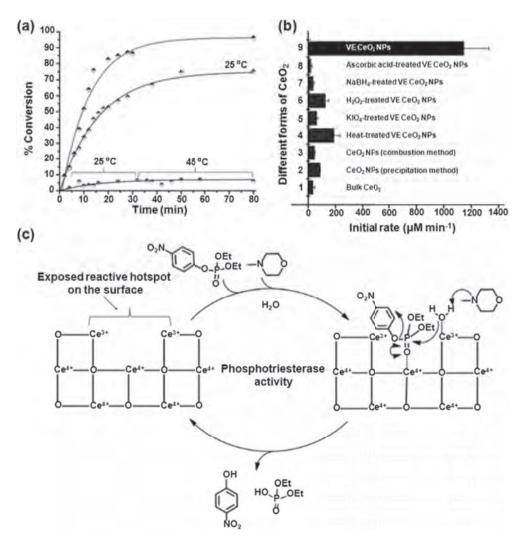


Fig. 34. (a) Comparison of initial rates of degradation of paraoxon by different forms of CeO<sub>2</sub> NPs. (b) Comparison of initial rates of various nanomaterials. (c) Proposed mechanism for the degradation of phosphotriesters by VE CeO<sub>2</sub> NPs. Adapted with permission from Ref. [220]. Copyright © 2016, WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim.

signaling [213]. Also, many chemical warfare agents are toxic organophosphorus compounds containing phosphonate ester bonds [214,215]. Thus, the phosphatase activity of natural and artificial enzymes has been widely improved for the rapid hydrolysis of these toxic chemicals. Previous studies showed that  $CeO_2$  nanostructures could mimic the phosphatase enzymes to cleave the phosphonate ester bonds [216–219]. Recent systematic dephosphorylation reaction kinetics demonstrated that the morphology and surface defects determined the catalytic activity of  $CeO_2$  nanozymes in an order of nanosphere > nano-octahedron > nanorod > nanocube based on their morphologies [218]. It is noted that the surface oxygen vacancy is recognized as the catalytic active site for the cleavage of phosphonate ester bonds. In this case, introducing more surface defects may be able to significantly enhance the phosphatase-mimetic activity of  $CeO_2$ .

The effect of surface defects in terms of surface Ce<sup>3+</sup> fractions was investigated by using the vacancy-engineered (VE) CeO<sub>2</sub> nanoparticles with the size of several nanometers [220]. As compared with other common CeO<sub>2</sub> nanoparticles, VE CeO<sub>2</sub> nanoparticles were synthesized in the presence of H<sub>2</sub>O<sub>2</sub>. A relative higher surface Ce<sup>3+</sup>/Ce<sup>4+</sup> fraction of 14.28% was then achieved for the VE CeO<sub>2</sub> nanoparticles. The synergistic activity between surface Ce<sup>3+</sup> and Ce<sup>4+</sup> could facilitate the hydrolysis of phosphonate ester bonds. In particular, the mimetic hydrolysis of paraoxon could be performed as a model reaction in weak basic solutions (pH ~ 10) with N-methylmorpholine as a base. The VE CeO2 nanoparticles were demonstrated to give the very high initial hydrolysis rate over  $1000 \,\mathrm{uM}\,\mathrm{min}^{-1}$  at  $45\,^{\circ}\mathrm{C}$  (Fig. 34a). In contrast, common CeO<sub>2</sub> nanoparticles and other metal oxides showed the poor catalytic activity for hydrolysis reactions (Fig. 34b). Interestingly, either an increase in the surface Ce<sup>3+</sup> fraction of VE CeO<sub>2</sub> nanoparticles by reductants (e.g., NaBH<sub>4</sub>, ascorbic acid) or a decrease in the value of artificial enzymes by oxidants (e.g., H2O2, KIO4), combining with thermal treatment in the presence of O2, would result in a sharp decrease in the mimetic activity for hydrolysis of paraoxon.

In details, catalytic mechanism investigations suggested that the surface Ce<sup>4+</sup> sites typically served as the adsorption sites for paraoxon molecules because of the polar nature of phosphoryl oxygen and the surface Ce<sup>3+</sup> species bind and activate H<sub>2</sub>O molecules. As shown in Fig. 34c, the cooperative surface Ce<sup>3+</sup> and Ce<sup>4+</sup> species facilitated the efficient nucleophilic attack of activated water or hydroxide at the phosphorus center. When the surface of CeO<sub>2</sub> catalysts was dominated by the Ce<sup>3+</sup> species, the surface of nanozymes was then mainly occupied by water molecules. Thus, the organophosphorus compounds became inaccessible to the catalyst surface, which led to degradation in the hydrolysis activity. Conversely, the surface of heavily oxidized CeO2 nanozymes allowed the selective adsorption of organophosphorus compounds and prevented the accessibility of water or hydroxide, which in turn reduced the phosphatase-mimetic activity. As a result, the optimized Ce<sup>3+</sup>/Ce<sup>4+</sup> ratio of CeO<sub>2</sub> was essential to induce the coadsorption of both reactants and to enable the mimetic hydrolysis activity.

#### 6. Conclusions and perspectives

In summary, the present review highlights the recent research developments in controlling the surface physicochemical properties of CeO<sub>2</sub> catalysts and their practical applications in advanced catalysis. It is evident that the catalytic activity of CeO<sub>2</sub> is boosted from the surface reversible oxidation states between Ce<sup>3+</sup> and Ce<sup>4+</sup>, which serves as the driving force for the redox chemistry, leads to the formation of the surface oxygen vacancy, and gives the unique surface bifunctional acid-base characteristics. Moreover, as the support of metal catalysts, manipulating the surface chemistry of

CeO<sub>2</sub> is also critical to the corresponding metal dispersion and size, catalytic activity and selectivity, and their capability for antisintering and anticoke formation. In general, surface properties of CeO<sub>2</sub> nanostructures can be effectively altered by different synthetic methods, various post-treatments, and surface modifications by small ions and molecules. Besides, the chemical redox etching and the synthetic pressure as well as the partial pressure of  $O_2$  in reaction vessels can also deliver operative and precise modulations of the surface chemical characteristics of CeO2 nanorods in a wide window. Benefiting from the controllable surface chemistry of CeO<sub>2</sub> catalysts, the catalytic efficiency and chemoselectivity can be well controlled for various catalytic reactions. From these observations, the recent progress and practical applications of heterogeneous CeO<sub>2</sub> catalysts are thoroughly discussed with the focus in CO oxidation, CO<sub>2</sub> conversion, organic synthesis, and their biomimetic performance.

At the same time, regulating the surface chemistry of  $CeO_2$  materials is revealed as a powerful and useful tool to tailor their catalytic selectivity and activity, and even to stimulate new catalytic phenomena evidenced by numerous studies mentioned in this review. Despite the significant progress has been achieved, understandings on the surface chemistry of  $CeO_2$  are still far from being resolved. In order to achieve the further insights, especially disclosing the detailed catalytic mechanism of surface regulated  $CeO_2$  at the molecular and/or atomic levels as well as seeking for the more practical catalytic applications, there remain more critical issues to be addressed.

- (1) Identifications of the spatial distribution of surface defects. Both the locations and concentrations of surface defects are observed to considerably affect the spatial and electronic structures of the catalytic active sites of CeO<sub>2</sub> catalysts. However, although the concentrations of surface defects might be assessed by various characterization techniques (e.g., XPS, and Raman), it is still difficult to evaluate the spatial locations of these species. For example, oxygen vacancies can be existed with many possible configurations, which are usually located on the surface, subsurface or even bulk, significantly affecting the electronic structures of corresponding surface/interface active sites in various ways. Also, the distance between these surface defects can considerably affect the catalytic activity of solid catalysts. It has been reported that CeO2 nanorods with clusters of oxygen vacancies could deliver the high catalytic activity for CO oxidation [221]. As a result, we can envision the importance of the spatial distribution of surface defects of a catalyst for specific catalytic reactions. Till now, this front has not been well explored because of the lack of sufficient technologies and synthetic strategies to manipulate the defects with welldefined configurations.
- (2) Catalytic performance of CeO<sub>2</sub> modified by small ions and molecules. The bifunctional acid-base properties of CeO<sub>2</sub> catalysts are widely recognized for their selective adsorption of small ions or molecules at the surface-specific sites of catalysts. Recently, the surface modifications of CeO<sub>2</sub> have been successfully demonstrated to further optimize the catalytic efficiency and selectivity for biomimetic reactions and organic catalysis [222]. In specific, the adsorption of small surface species can effectively regulate or even completely change the surface properties of solid catalysts in at least three aspects: (a) the re-distribution of the surface charge, (b) the modulation of the absorption of reactants and desorption of products, and (c) the construction of new surface catalytic sites. However, the effective surface area would get reduced because of the adsorbed species, as

compared with that of the "bare" ones. Therefore, it is highly important to find a descriptor to balance the surface modification and the available surface area. Along this direction, the current progress on modifying the surface chemistry of CeO<sub>2</sub> catalysts with small species remains quite limited. We need to understand the mechanism at the atomic level for surface modifications and make the adsorbed species cope with the existing surface species of CeO<sub>2</sub> catalysts in order further to promote their catalytic activity, selectivity, and stability. Moreover, surface modifications by large molecules, such as polymers and biological molecules, should also be considered, which can then create a microenvironment for specific reactions.

- (3) Precise controllability of surface properties of CeO<sub>2</sub>. The surface composition and electronic structure of CeO<sub>2</sub> nanomaterials are more complex than that we can imagine. Therefore, an atomic "observation" of the CeO<sub>2</sub> surface before any surface control is important. However, the residue ions adsorbed on the CeO<sub>2</sub> surface from wet chemical synthesis and the surface adsorbed gases (e.g., H<sub>2</sub>O, and CO<sub>2</sub>) after exposure to ambient conditions are difficult to be removed. They would inevitably add the uncertainty for the precise surface control to realize the desired surface catalytic configurations for advanced catalysis. In this case, it is essential to develop additional novel techniques to recognize the surface species accurately and to manipulate the surface chemistry precisely.
- (4) In-situ and operando characterizations of CeO<sub>2</sub> catalysts under the reaction conditions to provide a structure-activity relation. Generally, nanostructured catalysts are mostly synthesized using wet chemistry approaches, in which the obtained composition/structure may not as perfect as expected. Under the operating catalytic conditions, nano-sized catalysts would probably undergo the surface reconstruction. For example, since CeO<sub>2</sub>-base catalysts have been widely used for various catalytic reactions, including oxidation, hydrogenation, and hydrolysis, etc., any change on their surface states under catalytic conditions are unavoidable. Nevertheless, the precise understanding on their evolutions correlated to the catalytic performance have not been well recognized. Thus, it is recommended to emphasize that the in-situ and operando characterization techniques are urgently required in order to deliver an unambiguous structure-activity correlation of CeO<sub>2</sub> catalysts for the further performance enhancement.
- (5) Development of the theoretical methodology. Theoretical calculations are always employed to predict the surface chemistry of solid catalysts and to provide the guidelines for the design of catalysts by using engineered catalytic sites. However, majority of the current theoretical calculations always model the catalyst surface in a simplified way without considering the realistic reaction environments and complicated surface states of the catalysts. All these are wide known as so-called the "modeling-gap." For example, there are too many possibilities to simulate the surface defects of CeO<sub>2</sub> by considering their spatial configurations. It would become even worse when the dopants or surface adsorbed molecules are involved. Moreover, the dynamic surface reconstruction of CeO<sub>2</sub> under the operative catalytic conditions is also not considered for most of the cases. Therefore, more advanced computational methodologies are needed to clarify minimize the modeling gas, especially to assess the effect of above factors on the surface chemistry of solid catalysts and subsequently their catalytic behaviors.

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#### References

- [1] J. Paier, C. Penschke, J. Sauer, Chem. Rev. 113 (2013) 3949.
- [2] T. Montini, M. Melchionna, M. Monai, P. Fornasiero, Chem. Rev. 116 (2016)
- [3] J. Beckers, G. Rothenberg, Green Chem. 12 (2010) 939.
- 4] C. Sun, H. Li, L. Chen, Energy Environ. Sci. 5 (2012) 8475.
- [5] K. Wu, L.D. Sun, C.H. Yan, Adv. Energy Mater. 6 (2016) 1600501.
- [6] L. Vivier, D. Duprez, ChemSusChem 3 (2010) 654.
- [7] R.J. Gorte, AIChE J. 56 (2010) 1126.
- 8] M. Mogensen, N.M. Sammes, G.A. Tompsett, Solid State Ionics 129 (2000) 63.
- [9] W. Huang, Y. Gao, Catal. Sci. Technol. 4 (2014) 3772.
- [10] D. Zhang, X. Du, L. Shi, R. Gao, Dalton Trans. 41 (2012) 14455.
- [11] A. Trovarelli, Catal. Rev. 38 (1996) 439.
- [12] W. Huang, Acc. Chem. Res. 49 (2016) 520.
- [13] A. Trovarelli, Catalysis by Ceria and Related Materials, World Scientific, 2002.
- [14] C. Sun, U. Stimming, J. Power Sources 171 (2007) 247.
- [15] X. Feng, D.C. Sayle, Z.L. Wang, M.S. Paras, B. Santora, A.C. Sutorik, T.X. Sayle, Y. Yang, Y. Ding, X. Wang, Science 312 (2006) 1504.
- [16] S. Armini, J. De Messemaeker, C. Whelan, M. Moinpour, K. Maex, J. Electrochem, For. Soc. 155 (2008) H653.
- [17] T. Morimoto, H. Tomonaga, A. Mitani, Thin Solid Films 351 (1999) 61.
- [18] J. Kašpar, P. Fornasiero, M. Graziani, Catal. Today 50 (1999) 285.
- [19] M. Haneda, T. Kaneko, N. Kamiuchi, M. Ozawa, Catal. Sci. Technol. 5 (2015) 1792
- [20] M. Boaro, M. Vicario, C. de Leitenburg, G. Dolcetti, A. Trovarelli, Catal. Today 77 (2003) 407.
- [21] A. Paulenova, S. Creager, J. Navratil, Y. Wei, J. Power Sources 109 (2002) 431.
- [22] Z. Xie, F. Xiong, D. Zhou, Energy Fuels 25 (2011) 2399.
- [23] M. Zaki, M. Hasan, L. Pasupulety, Langmuir 17 (2001) 768.
- 24] T. Na, J. Liu, S. Wenjie, Chin. J. Catal. 34 (2013) 838.
- [25] E. Grulke, K. Reed, M. Beck, X. Huang, A. Cormack, S. Seal, Environ. Sci. Nano 1 (2014) 429.
- [26] R. Sakla, R. Hemamalini, K. Pranaw, S. Kumar Khare, Curr. Bionanotechnol 2 (2016) 122.
- [27] C. Walkey, S. Das, S. Seal, J. Erlichman, K. Heckman, L. Ghibelli, E. Traversa, J.F. McGinnis, W.T. Self, Environ. Sci. Nano 2 (2015) 33.
- [28] S. Sato, R. Takahashi, M. Kobune, H. Gotoh, Appl. Catal. A Gen. 356 (2009) 57.
- 29] W. Huang, Top. Catal. 56 (2013) 1363.
- [30] D.R. Mullins, P.M. Albrecht, F. Calaza, Top. Catal. 56 (2013) 1345.
- [31] S.D. Senanayake, D. Stacchiola, J.A. Rodriguez, Acc. Chem. Res. 46 (2013) 1702.
- [32] S. Song, X. Wang, H. Zhang, NPG Asia Mater. 7 (2015) e179.
- [33] J.A. Rodriguez, D.C. Grinter, Z. Liu, R.M. Palomino, S.D. Senanayake, Chem. Soc. Rev. 46 (2017) 1824.
- [34] J.F. Jerratsch, X. Shao, N. Nilius, H.J. Freund, C. Popa, M.V. Ganduglia-Pirovano, A.M. Burow, J. Sauer, Phys. Rev. Lett. 106 (2011) 246801.
- [35] C.T. Campbell, C.H. Peden, Science 309 (2005) 713.
- [36] B.W. Sheldon, V.B. Shenoy, Phys. Rev. Lett. 106 (2011) 216104.
- [37] J. Li, Z. Zhang, Z. Tian, X. Zhou, Z. Zheng, Y. Ma, Y. Qu, J. Mater. Chem. A 2 (2014) 16459.
- [38] Q. Yuan, H.H. Duan, L.L. Li, L.D. Sun, Y.W. Zhang, C.H. Yan, J. Colloid Interface Sci. 335 (2009) 151.
- [39] J.Y. Park, L.R. Baker, G.A. Somorjai, Chem. Rev. 115 (2015) 2781.
- [40] J.B. Park, J. Graciani, J. Evans, D. Stacchiola, S. Ma, P. Liu, A. Nambu, J.F. Sanz, J. Hrbek, J.A. Rodriguez, Proc. Natl. Acad. Sci. USA 106 (2009) 4975.
- [41] K. Yoon, Y. Yang, P. Lu, D. Wan, H.C. Peng, K. Stamm Masias, P.T. Fanson, C.T. Campbell, Y. Xia, Angew. Chem. Int. Ed. 51 (2012) 9543.
- [42] D. Kong, G. Wang, Y. Pan, S. Hu, J. Hou, H. Pan, C.T. Campbell, J. Zhu, J. Phys. Chem. C 115 (2011) 6715.
- [43] D.R. Mullins, Surf. Sci. Rep. 70 (2015) 42.
- [44] W. Huang, G. Sun, T. Cao, Chem. Soc. Rev. 46 (2017) 1977.
- [45] M.A. Henderson, C. Perkins, M.H. Engelhard, S. Thevuthasan, C.H. Peden, Surf. Sci. 526 (2003) 1.
- [46] K. Deori, D. Gupta, B. Saha, S. Deka, ACS Catal. 4 (2014) 3169.
- [47] H. Yao, Y.Y. Yao, J. Catal. 86 (1984) 254.
- [48] D. Duprez, C. Descorme, T. Birchem, E. Rohart, Top. Catal. 16 (2001) 49.
- [49] Y. Madier, C. Descorme, A. Le Govic, D. Duprez, J. Phys. Chem. B 103 (1999) 10999.
- [50] P. Fornasiero, R. Dimonte, G.R. Rao, J. Kaspar, S. Meriani, A. Trovarelli, M. Graziani, J. Catal. 151 (1995) 168.
- [51] T. Murota, T. Hasegawa, S. Aozasa, H. Matsui, M. Motoyama, J. Alloy. Comp.

- 193 (1993) 298.
- [52] T. Miki, T. Ogawa, M. Haneda, N. Kakuta, A. Ueno, S. Tateishi, S. Matsuura, M. Sato, J. Phys. Chem. 94 (1990) 6464.
- [53] T. Maillet, Y. Madier, R. Taha, J. Barbier, D. Duprez, Stud. Surf. Sci. Catal. 112 (1997) 267.
- [54] B.K. Cho, J. Catal. 131 (1991) 74.
- [55] E. Su, C. Montreuil, W. Rothschild, Appl. Catal 17 (1985) 75.
- [56] J.T. Kummer, Prog. Energ. Combust. 6 (1980) 177.
- [57] X.F. Yang, A. Wang, B. Qiao, J. Li, J. Liu, T. Zhang, Acc. Chem. Res. 46 (2013) 1740
- [58] M. Yang, S. Li, Y. Wang, J.A. Herron, Y. Xu, L.F. Allard, S. Lee, J. Huang, M. Mavrikakis, M. Flytzani-Stephanopoulos, Science 346 (2014) 1498.

- [59] S. Liang, C. Hao, Y. Shi, ChemCatChem 7 (2015) 2559.
  [60] C.R. Henry, Surf. Sci. Rep. 31 (1998) 231.
  [61] V. Polshettiwar, R.S. Varma, Green Chem. 12 (2010) 743.
- [62] J.K. Norskov, T. Bligaard, B. Hvolbaek, F. Abild-Pedersen, I. Chorkendorff, C.H. Christensen, Chem. Soc. Rev. 37 (2008) 2163.
- [63] S. Schauermann, N. Nilius, S. Shaikhutdinov, H.J. Freund, Acc. Chem. Res. 46 (2013) 1673
- [64] M. Hirano, E. Kato, I. Mater, Sci. Lett. 15 (1996) 1249.
- [65] A.I.Y. Tok, F.Y.C. Boey, Z. Dong, X. Sun, J. Mater. Process. Technol. 190 (2007)
- [66] M. Hirano, E. Kato, J. Am. Ceram. Soc. 82 (1999) 786.
- M. Hirano, E. Kato, J. Am. Ceram. Soc. 79 (1996) 777.
- [68] H.X. Mai, L.D. Sun, Y.W. Zhang, R. Si, W. Feng, H.P. Zhang, H.C. Liu, C.H. Yan, J. Phys. Chem. B 109 (2005) 24380.
- [69] G. Shen, Q. Wang, Z. Wang, Y. Chen, Mater. Lett. 65 (2011) 1211.
- L. Yan, R. Yu, C. Jun, X. Xianran, Growth Des. 8 (2008) 1474.
- [71] F. Lin, D.T. Hoang, C.K. Tsung, W. Huang, S.H.Y. Lo, J.B. Wood, H. Wang, J. Tang, P. Yang, Nano Res. 4 (2010) 61.
- [72] Z. Guo, F. Du, G. Li, Z. Cui, Inorg. Chem. 45 (2006) 4167.
- [73] Q. Wu, F. Zhang, P. Xiao, H. Tao, X. Wang, Z. Hu, Y. Lu, J. Phys. Chem. C 112 (2008) 17076.
- [74] C. Sun, J. Sun, G. Xiao, H. Zhang, X. Qiu, H. Li, L. Chen, J. Phys. Chem. B 110 (2006) 13445.
- J.G. Caporaso, J. Kuczynski, J. Stombaugh, K. Bittinger, F.D. Bushman, E.K. Costello, N. Fierer, A.G. Peña, J.K. Goodrich, J.I. Gordon, Br. J. Pharmacol. 7 (2010) 335.
- [76] S. Yang, L. Gao, J. Am. Chem. Soc. 128 (2006) 9330.
- X. Wang, Z. Jiang, B. Zheng, Z. Xie, L. Zheng, CrystEngComm 14 (2012) 7579.
- [78] D. Wang, Y. Kang, V. Doan-Nguyen, J. Chen, R. Küngas, N.L. Wieder, K. Bakhmutsky, R.J. Gorte, C.B. Murray, Angew. Chem. Int. Ed. 50 (2011) 4378.
- S. Laha, R. Ryoo, Chem. Commun. (2003) 2138.
- [80] W. Yuejuan, M. Jingmeng, L. Mengfei, F. Ping, H. Mai, J. Rare Earth 25 (2007)
- [81] C. Sun, H. Li, Z. Wang, L. Chen, X. Huang, Chem. Lett. 33 (2004) 662.
- [82] L. González-Rovira, J.M. Sánchez-Amaya, M. López-Haro, E. del Rio, A.B. Hungría, P. Midgley, J.J. Calvino, S. Bernal, F.J. Botana, Nano Lett. 9 (2009)
- [83] R.J. La, Z.A. Hu, H.L. Li, X.L. Shang, Y.Y. Yang, Mat. Sci. Eng. A Struct. 368 (2004) 145.
- Y. Chen, J. Lu, J. Porous Mater. 19 (2012) 289.
- Y. Sun, Q. Liu, S. Gao, H. Cheng, F. Lei, Z. Sun, Y. Jiang, H. Su, S. Wei, Y. Xie, Nat. Commun. 4 (2013) 2899.
- [86] E. Mamontov, T. Egami, R. Brezny, M. Koranne, S. Tyagi, J. Phys. Chem. B 104 (2000) 11110.
- [87] N. Shehata, K. Meehan, I. Hassounah, M. Hudait, N. Jain, M. Clavel, S. Elhelw, N. Madi, Nanoscale Res. Lett. 9 (2014) 1.
- [88] N.J. Lawrence, J.R. Brewer, L. Wang, T.S. Wu, J. Wells-Kingsbury, M.M. Ihrig, G. Wang, Y.L. Soo, W.N. Mei, C.L. Cheung, Nano Lett. 11 (2011) 2666.
- [89] T.S. Sakthivel, D.L. Reid, U.M. Bhatta, G. Möbus, D.C. Sayle, S. Seal, Nanoscale 7 (2015) 5169.
- [90] M.M. Khan, S.A. Ansari, D. Pradhan, D.H. Han, J. Lee, M.H. Cho, Ind. Eng. Chem. Res. 53 (2014) 9754.
- [91] Y.J. Wang, H. Dong, G.M. Lyu, H.Y. Zhang, J. Ke, L.Q. Kang, J.L. Teng, L.D. Sun, R. Si, J. Zhang, Nanoscale 7 (2015) 13981.
- [92] W. Gao, Z. Zhang, J. Li, Y. Ma, Y. Qu, Nanoscale 7 (2015) 11686.
- [93] J. Li, Z. Zhang, W. Gao, S. Zhang, Y. Ma, Y. Qu, ACS Appl. Mater. Interfaces 8 (2016) 22988.
- [94] K. Liu, A. Wang, T. Zhang, ACS Catal. 2 (2012) 1165.
- [95] S. Chen, L. Luo, Z. Jiang, W. Huang, ACS Catal. 5 (2015) 1653.
- [96] Z. Hu, X. Liu, D. Meng, Y. Guo, Y. Guo, G. Lu, ACS Catal. 6 (2016) 2265.
- [97] Y. Gao, W. Wang, S. Chang, W. Huang, ChemCatChem 5 (2013) 3610.
- [98] K. An, S. Alayoglu, N. Musselwhite, S. Plamthottam, G.R.M. Melaet, A.E. Lindeman, G.A. Somorjai, J. Am. Chem. Soc. 135 (2013) 16689.
- [99] S.L. Zhong, L.F. Zhang, L. Wang, W.X. Huang, C.M. Fan, A.W. Xu, J. Phys. Chem. C 116 (2012) 13127.
- [100] J. Qin, J. Lu, M. Cao, C. Hu, Nanoscale 2 (2010) 2739.
- [101] J. Li, P. Zhu, R. Zhou, J. Power Sources 196 (2011) 9590.
- [102] D.M. Kempaiah, S. Yin, T. Sato, CrystEngComm 13 (2011) 741.
- [103] S. Chang, M. Li, Q. Hua, L. Zhang, Y. Ma, B. Ye, W. Huang, J. Catal. 293 (2012) 195.
- [104] Z. Wu, M. Li, S.H. Overbury, J. Catal. 285 (2012) 61.
- [105] C. Ho, J.C. Yu, T. Kwong, A.C. Mak, S. Lai, Chem. Mater. 17 (2005) 4514.
   [106] C. Pan, D. Zhang, L. Shi, J. Fang, Eur. J. Inorg. Chem. 2008 (2008) 2429.

- [107] Z. Yang, J. Wei, H. Yang, L. Liu, H. Liang, Y. Yang, Eur. J. Inorg. Chem. 2010 (2010) 3354.
- [108] A.P. Amrute, C. Mondelli, M. Moser, G. Novell-Leruth, N. López, D. Rosenthal, R. Farra, M.E. Schuster, D. Teschner, T. Schmidt, J. Pérez-Ramírez, J. Catal. 286 (2012) 287.
- [109] R. Farra, M. Eichelbaum, R. Schlögl, L. Szentmiklósi, T. Schmidt, A.P. Amrute, C. Mondelli, J. Pérez-Ramírez, D. Teschner, J. Catal. 297 (2013) 119.
- [110] C. Li, Y. Sun, I. Djerdj, P. Voepel, C.C. Sack, T. Weller, R. Ellinghaus, J. Sann, Y. Guo, B.M. Smarsly, H. Over, ACS Catal, 7 (2017) 6453.
- [111] Z. Yang, D. Han, D. Ma, H. Liang, L. Liu, Y. Yang, Cryst. Growth Des. 10 (2009) 291.
- [112] G. Hua, L. Zhang, G. Fei, M. Fang, J. Mater. Chem. 22 (2012) 6851.
- [113] C.H. Huang, C.S. Tan, Aerosol Air Qual. Res. 14 (2014) 480.
- [114] Y. Li, X. Cui, K. Dong, K. Junge, M. Beller, ACS Catal. 7 (2017) 1077.
- [115] J. Klankermayer, S. Wesselbaum, K. Beydoun, W. Leitner, Angew. Chem. Int. Ed. 55 (2016) 7296.
- [116] G. Centi, E.A. Quadrelli, S. Perathoner, Energy Environ. Sci. 6 (2013) 1711.
  [117] M. Aresta, A. Dibenedetto, A. Angelini, Chem. Rev. 114 (2014) 1709.
- [118] Q. Liu, L. Wu, R. Jackstell, M. Beller, Nat. Commun. 6 (2015) 5933.
- [119] M. Tamura, M. Honda, Y. Nakagawa, K. Tomishige, J. Chem. Technol. Biotechnol 89 (2014) 19
- [120] B. Santos, C. Pereira, V. Silva, J. Loureiro, A. Rodrigues, Appl. Catal. A Gen. 455 (2013) 219.
- [121] M. Honda, M. Tamura, K. Nakao, K. Suzuki, Y. Nakagawa, K. Tomishige, ACS Catal. 4 (2014) 1893.
- [122] M. Honda, M. Tamura, Y. Nakagawa, K. Tomishige, Catal. Sci. Technol. 4 (2014) 2830.
- [123] M. Honda, S. Kuno, S. Sonehara, K.i. Fujimoto, K. Suzuki, Y. Nakagawa, K. Tomishige, ChemCatChem 3 (2011) 365.
- [124] M. Honda, M. Tamura, Y. Nakagawa, S. Sonehara, K. Suzuki, K.i. Fujimoto, K. Tomishige, ChemSusChem 6 (2013) 1341.
- [125] A. Bansode, A. Urakawa, ACS Catal. 4 (2014) 3877.
- [126] Y. Yoshida, Y. Arai, S. Kado, K. Kunimori, K. Tomishige, Catal. Today 115 (2006) 95.
- [127] S. Wang, L. Zhao, W. Wang, Y. Zhao, G. Zhang, X. Ma, J. Gong, Nanoscale 5 (2013) 5582.
- [128] J. Xu, K.Z. Long, F. Wu, B. Xue, Y.X. Li, Y. Cao, Appl. Catal. A Gen. 484 (2014) 1. [129] E. Leino, P. Mäki-Arvela, V. Eta, N. Kumar, F. Demoisson, A. Samikannu,
- A.R. Leino, A. Shchukarev, D.Y. Murzin, J.P. Mikkola, Catal. Today 210 (2013)
- [130] E. Leino, N. Kumar, P. Mäki-Arvela, A. Aho, K. Kordás, A.R. Leino, A. Shchukarev, D.Y. Murzin, J.P. Mikkola, Mater. Chem. Phys. 143 (2013) 65.
- [131] K. Reed, A. Cormack, A. Kulkarni, M. Mayton, D. Sayle, F. Klaessig, B. Stadler, Environ. Sci. Nano 1 (2014) 390.
- [132] R.K. Grasselli, Top. Catal. 15 (2001) 93.
- [133] W.C. Chueh, S.M. Haile, ChemSusChem 2 (2009) 735.
- [134] W.C. Chueh, C. Falter, M. Abbott, D. Scipio, P. Furler, S.M. Haile, A. Steinfeld, Science 330 (2010) 1797.
- [135] B. Wei, R. Fakhrai, B. Saadatfar, Int. J. Hydrogen Energy 39 (2014) 12353.
- [136] Y. Wang, B. Li, C. Zhang, L. Cui, S. Kang, X. Li, L. Zhou, Appl. Catal. B Environ. 130 (2013) 277.
- [137] D. Jiang, W. Wang, E. Gao, S. Sun, L. Zhang, Chem. Commun. 50 (2014) 2005.
- [138] J. Jiao, Y. Wei, Z. Zhao, J. Liu, J. Li, A. Duan, G. Jiang, Ind. Eng. Chem. Res. 53 (2014) 17345.
- [139] X. Yang, S. Kattel, S.D. Senanayake, J.A. Boscoboinik, X. Nie, J.S. Graciani, J.A. Rodriguez, P. Liu, D.J. Stacchiola, J.G. Chen, J. Am. Chem. Soc. 137 (2015) 10104.
- [140] P. Li, Y. Zhou, Z. Zhao, Q. Xu, X. Wang, M. Xiao, Z. Zou, J. Am. Chem. Soc. 137 (2015) 9547.
- [141] P. Kumar, A. Kumar, C. Joshi, R. Singh, S. Saran, S.L. Jain, RSC Adv. 5 (2015) 42414.
- [142] A. Primo, T. Marino, A. Corma, R. Molinari, H. Garcia, J. Am. Chem. Soc. 133 (2011) 6930.
- [143] X. Zhang, R. You, D. Li, T. Cao, W. Huang, ACS Appl. Mater. Interfaces 9 (2017) 35897.
- [144] R. You, X. Zhang, L. Luo, Y. Pan, H. Pan, J. Yang, L. Wu, X. Zheng, Y. Jin, W. Huang, J. Catal. 348 (2017) 189.
- [145] Y. Liu, L. Luo, Y. Gao, W. Huang, Appl. Catal. B Environ. 197 (2016) 214.
- [146] Y. Wang, F. Wang, Q. Song, Q. Xin, S. Xu, J. Xu, J. Am. Chem. Soc. 135 (2013)
- [147] J.C. Westfahl, T.L. Gresham, J. Am. Chem. Soc. 77 (1955) 936.
- [148] M. Tamura, A. Satsuma, K.I. Shimizu, Catal. Sci. Technol 3 (2013) 1386.
- [149] M. Tamura, K. Sawabe, K. Tomishige, A. Satsuma, K.I. Shimizu, ACS Catal. 5 (2015) 20.
- [150] A. Rapeyko, M.J. Climent, A. Corma, P. Concepción, S. Iborra, ACS Catal. 6 (2016) 4564.
- [151] J. Liu, X.P. Wu, S. Zou, Y. Dai, L. Xiao, X.Q. Gong, J. Fan, J. Phys. Chem. C 118 (2014) 24950.
- [152] M. Tamura, K. Tomishige, Angew. Chem. Int. Ed. 54 (2015) 864.
- [153] A. Grirrane, A. Corma, H. García, Science 322 (2008) 1661.
- [154] B. Li, T. Gu, T. Ming, J. Wang, P. Wang, J. Wang, J.C. Yu, ACS Nano 8 (2014) 8152.
- [155] B. Zhang, S. Kaziz, H. Li, M.G. Hevia, D. Wodka, C. Mazet, T. Bürgi, N. Barrabés, J. Phys. Chem. C 119 (2015) 11193.
- [156] A. Primo, T. Marino, A. Corma, R. Molinari, H. Garcia, J. Am. Chem. Soc. 133

(2011) 6930.

- [157] P. Zhang, H. Lu, Y. Zhou, L. Zhang, Z. Wu, S. Yang, H. Shi, Q. Zhu, Y. Chen, S. Dai, Nat. Commun. 6 (2015) 8446.
- [158] J. Lv, Y. Shen, L. Peng, X. Guo, W. Ding, Chem. Commun. 46 (2010) 5909.
- [159] K. Deori, D. Gupta, B. Saha, S. Deka, ACS Catal. 4 (2014) 3169.
- [160] A. Leyva-Pérez, D. Cómbita-Merchán, J.R. Cabrero-Antonino, S.I. Al-Resayes, A. Corma, ACS Catal. 3 (2013) 250.
- [161] M. Tamura, R. Kishi, Y. Nakagawa, K. Tomishige, Nat. Commun. 6 (2015) 8580
- [162] S. Zhang, J. Li, Z. Xia, C. Wu, Z. Zhang, Y. Ma, Y. Qu, Nanoscale 9 (2017) 3140.
  [163] S. Zhang, C.R. Chang, Z.Q. Huang, J. Li, Z. Wu, Y. Ma, Z. Zhang, Y. Wang, Y. Qu, J. Am. Chem. Soc. 138 (2016) 2629.
- [164] S. Zhang, Z. Xia, T. Ni, H. Zhang, C. Wu, Y. Qu, J. Mate. Chem. A 5 (2017) 3260.
- [165] N. Perret, X. Wang, J.J. Delgado, G. Blanco, X. Chen, C.M. Olmos, S. Bernal, M.A. Keane, J. Catal. 317 (2014) 114.
- [166] S.D. Senanayake, P.J. Ramírez, I. Waluyo, S. Kundu, K. Mudiyanselage, Z. Liu, Z. Liu, S. Axnanda, D.J. Stacchiola, J. Evans, J.A. Rodriguez, J. Phys. Chem. C 120  $(2016)\ 1778$
- [167] M. Tamura, T. Kitanaka, Y. Nakagawa, K. Tomishige, ACS Catal. 6 (2016) 376.
- [168] G. Vicario, G. Balducci, S. Fabris, S. de Gironcoli, S. Baroni, J. Phys. Chem. B 110 2006) 19380.
- [169] M. Chong, D.G. Cheng, L. Liu, F. Chen, X. Zhan, Catal. Lett. 114 (2007) 198.
- [170] D.G. Cheng, M. Chong, F. Chen, X. Zhan, Catal. Lett. 120 (2007) 82.
- [171] G. Vile, B. Bridier, J. Wichert, J. Perez-Ramirez, Angew. Chem. Int. Ed. 51 (2012) 8620.
- [172] G. Vile, S. Colussi, F. Krumeich, A. Trovarelli, J. Perez-Ramirez, Angew. Chem. Int. Ed. 53 (2014) 12069.
- [173] M. García-Melchor, L. Bellarosa, N. López, ACS Catal, 4 (2014) 4015.
- [174] S. Zhang, Z.Q. Huang, Y. Ma, W. Gao, J. Li, F.X. Cao, L. Li, C.R. Chang, Y. Qu, Nat. Commun. 8 (2017) 15266.
- [175] H.Z. Zhu, Y.M. Lu, F.J. Fan, S.H. Yu, Nanoscale 5 (2013) 7219.
- [176] Z.Q. Huang, L.P. Liu, S. Qi, S. Zhang, Y. Qu, C.R. Chang, ACS Catal. (2017) 546.[177] M. Tamura, S.M.A. Hakim Siddiki, K.I. Shimizu, Green Chem. 15 (2013) 1641.
- [178] Z. Zhang, Y. Wang, J. Lu, C. Zhang, M. Wang, M. Li, X. Liu, F. Wang, ACS Catal. 6 (2016) 8248.
- [179] E.G. Heckert, A.S. Karakoti, S. Seal, W.T. Self, Biomaterials 29 (2008) 2705.
- [180] K.L. Heckman, W. DeCoteau, A. Estevez, K.J. Reed, W. Costanzo, D. Sanford, J.C. Leiter, J. Clauss, K. Knapp, C. Gomez, ACS Nano 7 (2013) 10582.
- [181] A. Karakoti, S. Singh, J.M. Dowding, S. Seal, W.T. Self, Chem. Soc. Rev. 39 (2010) 4422.
- [182] S. Das, J.M. Dowding, K.E. Klump, J.F. McGinnis, W. Self, S. Seal, Nanomedicine 8 (2013) 1483.
- [183] Z. Tian, J. Li, Z. Zhang, W. Gao, X. Zhou, Y. Qu, Biomaterials 59 (2015) 116.
- [184] C. Xu, X. Qu, NPG Asia Mater. 6 (2014) e90.
- [185] C.K. Kim, T. Kim, I.Y. Choi, M. Soh, D. Kim, Y.J. Kim, H. Jang, H.S. Yang, J.Y. Kim, H.K. Park, Angew. Chem. Int. Ed. 51 (2012) 11039.
- [186] S.S. Lee, W. Song, M. Cho, H.L. Puppala, P. Nguyen, H. Zhu, L. Segatori, V.L. Colvin, ACS Nano 7 (2013) 9693.
- [187] F. Pagliari, C. Mandoli, G. Forte, E. Magnani, S. Pagliari, G. Nardone, S. Licoccia, M. Minieri, P. Di Nardo, E. Traversa, ACS Nano 6 (2012) 3767.
- [188] P. Xu, B. Maidment 3rd, V. Antonic, I. Jackson, S. Das, A. Zodda, X. Zhang,

- S. Seal, Z. Vujaskovic, Radiat. Res. 185 (2016) 516.
- [189] H. Wei, E. Wang, Chem. Soc. Rev. 42 (2013) 6060.
- [190] J. Chen, S. Patil, S. Seal, J.F. McGinnis, Nat. Nanotechnol. 1 (2006) 142.
- [191] B. Liu, J. Liu, Nano Res. (2017) 1.
- [192] G. Song, Y. Chen, C. Liang, X. Yi, J. Liu, X. Sun, S. Shen, K. Yang, Z. Liu, Adv. Mater. 28 (2016) 7143.
- [193] C. Zhang, W. Bu, D. Ni, S. Zhang, Q. Li, Z. Yao, J. Zhang, H. Yao, Z. Wang, J. Shi, Angew. Chem. Int. Ed. 55 (2016) 2101.
- [194] H. Li, C. Liu, Y.P. Zeng, Y.H. Hao, J.-W. Huang, Z.Y. Yang, R. Li, ACS Appl, Mater. Interfaces 8 (2016) 31510.
- [195] S.M. Hirst, A.S. Karakoti, R.D. Tyler, N. Sriranganathan, S. Seal, C.M. Reilly. Small 5 (2009) 2848.
- [196] E.J. Park, J. Choi, Y.K. Park, K. Park, Toxicology 245 (2008) 90.
- [197] I Celardo LZ Pedersen F Traversa I Ghibelli Nanoscale 3 (2011) 1411
- [198] X. Wu, Y. Zhang, Y. Lu, S. Pang, K. Yang, Z. Tian, Y. Pei, Y. Qu, F. Wang, Z. Pei, I. Mater. Chem. B 5 (2017) 3483.
- [199] C. Korsvik, S. Patil, S. Seal, W.T. Self, Chem. Commun. (2007) 1056.
   [200] T. Xia, M. Kovochich, M. Liong, L. Mädler, B. Gilbert, H. Shi, J.I. Yeh, J.I. Zink, A.E. Nel, ACS Nano 2 (2008) 2121.
- [201] B.K. Pierscionek, Y. Li, A.A. Yasseen, L.M. Colhoun, R.A. Schachar, W. Chen,
- Nanotechnology 21 (2009) 035102. [202] L. Gao, J. Zhuang, L. Nie, J. Zhang, Y. Zhang, N. Gu, T. Wang, J. Feng, D. Yang, S. Perrett, Nat. Nanotechnol. 2 (2007) 577.
- [203] A.K. Yagati, T. Lee, J. Min, J.W. Choi, Biosens. Bioelectron. 47 (2013) 385.
- [204] Y. Yang, Z. Mao, W. Huang, L. Liu, J. Li, J. Li, Q. Wu, Sci. Rep. 6 (2016).
  [205] X. Jiao, H. Song, H. Zhao, W. Bai, L. Zhang, Y. Lv, Anal. Methods 4 (2012) 3261.
- [206] F. Caputo, M. De Nicola, A. Sienkiewicz, A. Giovanetti, I. Bejarano, S. Licoccia, E. Traversa, L. Ghibelli, Nanoscale 7 (2015) 15643.
- [207] S. Singh, T. Dosani, A.S. Karakoti, A. Kumar, S. Seal, W.T. Self, Biomaterials 32 (2011) 6745.
- [208] A. Asati, S. Santra, C. Kaittanis, S. Nath, J.M. Perez, Angew. Chem. Int. Ed. 48 (2009) 2308.
- [209] M. Ornatska, E. Sharpe, D. Andreescu, S. Andreescu, Anal. Chem. 83 (2011) 4273
- [210] A. Hayat, J. Cunningham, G. Bulbul, S. Andreescu, Anal. Chim. Acta 885 (2015) 140.
- [211] B. Liu, Z. Huang, J. Liu, Nanoscale 8 (2016) 13562.
- [212] X. Li, M. Wilmanns, J. Thornton, M. Kohn, Sci. Signal. 6 (2013) 10.
- [213] F. Sacco, L. Perfetto, L. Castagnoli, G. Cesareni, FEBS Lett. 586 (2012) 2732.
- [214] D.M. Maxwell, K.M. Brecht, I. Koplovitz, R.E. Sweeney, Arch. Toxicol. 80 (2006) 756.
- [215] M. Eddleston, N.A. Buckley, P. Eyer, A.H. Dawson, Lancet 371 (2008) 597.
- [216] M.H. Kuchma, C.B. Komanski, J. Colon, A. Teblum, A.E. Masunov, B. Alvarado, S. Babu, S. Seal, J. Summy, C.H. Baker, Nanomed-Nanotechnol 6 (2010) 738.
- [217] A.A. Vernekar, T. Das, G. Mugesh, Angew. Chem. Int. Ed. 55 (2016) 1412.
- [218] M.J. Manto, P. Xie, C. Wang, ACS Catal. 7 (2017) 1931.
- [219] B. Fatima, M. Najam-ul-Haq, F. Jabeen, S. Majeed, M.N. Ashiq, S.G. Musharraf, M.A. Shad, G. Xu, Analyst 138 (2013) 5059.
- [220] A.A. Vernekar, T. Das, G. Mugesh, Angew. Chem. Int. Ed. 55 (2016) 1412.
- [221] X. Liu, K. Zhou, L. Wang, B. Wang, Y. Li, J. Am. Chem. Soc. 131 (2009) 3140.
- [222] B. Liu, J. Liu, Nano Res. 10 (2017) 1125.