

MICROELECTRONIC TECHNOLOGIES

EXAMS NOTES

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June 20, 2017

...Scooby-doo?...

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Chapter 1

Exams notes

1.1 Ion implant

1.1.1 Stopping mechanism

Two main mechanisms are in action in ion stopping

- $S_n(E)$ Two body collision with other atoms.
Can be modelled as a Coulomb scattering with a correcting factor for the atom shielding.
It becomes relevant only at the end of the process.
- $S_e(E)$ Dragged by electrons (electronic stopping power).
Polarization fields created by stationary charge atoms create a viscous space where the ion has to travel. Momentum is exchanged with orbital electrons.
The energy can be modelled as a motion into a viscous medium

$$S_e(E) = k\sqrt{E} \quad (1.1)$$

The dispersion relation with energy is

$$\frac{dE}{dx} = -N(S_n(E) + S_e(E)) \quad (1.2)$$

That integrated gives us the projected range as

$$R_p = \int_0^{R_p} dx = \int_0^{E_{\text{implant}}} \frac{dE}{N(S_n(E) + S_e(E))} \quad (1.3)$$

1.1.2 Range and +1 model

Given an implant of an element with a dose Q and an energy E we can find from tables the values of

- R_p the projected range of our implant that is the maximum of our gaussian distribution that we expect
- ΔR_p the standard deviation of our gaussian distribution

So we will have a distribution like

$$C(x) = C_p \exp\left(-\frac{(x - R_p)^2}{2\Delta R_p^2}\right) \quad (1.4)$$

where C_p is the concentration at the peak of our distribution.

The total dose implanted will be $Q = \sqrt{2\pi\Delta R_p^2} C_p$ and so $C_p = \frac{Q}{\sqrt{2\pi\Delta R_p^2}}$ doing so we can write

$$C(x) = \frac{Q}{\sqrt{2\pi\Delta R_p^2}} \exp\left(-\frac{(x - R_p)^2}{2\Delta R_p^2}\right) \quad (1.5)$$

If we want to know the dose that is present after a certain depth x

$$Q_{imp} = \int_x^{+\infty} C_p \exp\left(-\frac{(x - R_p)^2}{2\Delta R_p^2}\right) dx \quad (1.6)$$

That becomes

$$Q_{imp} = \frac{Q}{2} \operatorname{erfc}\left(\frac{x - R_p}{\sqrt{2}\Delta R_p}\right) \quad (1.7)$$

The function erfc has the following proprieties

- $\operatorname{erfc}(x) = 1 - \operatorname{erf}(x)$ if $x > 0$
- $\operatorname{erfc}(x) = 1 + \operatorname{erf}(x)$ if $x < 0$

If we want to estimate the amount of interstitial caused by an implant using the +1 model we can say that the concentration of interstitial is equal to the dose Q implanted in silicon. If the silicon is fully amorphised and then re-crystallized there are no interstitial due to ion implantation since in an amorph crystal do not exists interstitials or defects.

1.1.3 With dopant diffusion

After an implant (with a certain dose Q and having also R_p and ΔR_p) there will always be a thermal annealing process for a certain time t at a temperature T .

The profile of the doping will be a gaussian with the standard deviation modified by the diffusion due to thermal treatment so like

$$C(x) = \frac{Q_i}{\sqrt{2\pi(\Delta R_p^2 + 2Dt)}} \exp\left(-\frac{(x - R_p)^2}{2(\Delta R_p^2 + 2Dt)}\right) \quad (1.8)$$

From this we can know the concentration of dopants for all x .

Beware to correctly set Q depending on the symmetry of the system.

1.1.4 Diffusion length of dopants and supersaturation of I

Diffusion length is defined as the sigma of the gaussian considering an annealing process in an inert ambient and not the ion implantation so

$$l_d = \sqrt{D^*t} \quad (1.9)$$

where D^* is the equilibrium diffusivity in an inert ambient and it has an Ahrrenius form so depends a lot on temperature.

In case we are in a oxidizing/nitride ambient we have an enhanced diffusivity that can be written as

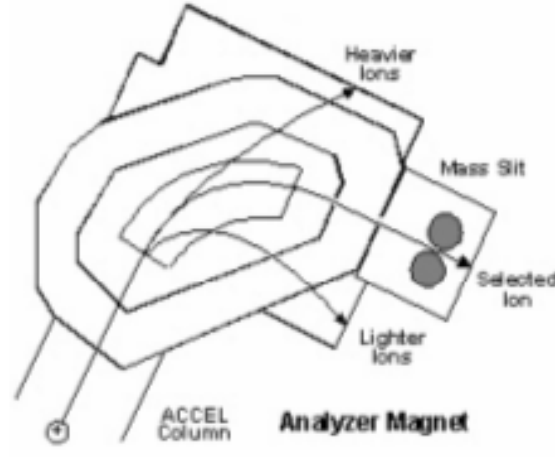
$$D_{eff} = D^* \left(\frac{C_I}{C_I^*} f_i + \frac{C_V}{C_V^*} f_v \right) \quad (1.10)$$

where we used

- $f_{i/v}$ interstitial/vacancy type mechanism fraction (tipically one is 1 one is 0)
- $C_{V/I}^*$ vacancy/interstitial equilibrium concentration
- $C_{V/I}$ vacancy/interstitial concentration

From this expression we can derive the supersaturation of I or V that is the term $\frac{C_I}{C_I^*}$

1.1.5 Analyser; tune for B



Electrons enters in the chamber with potential V where a there is a perpendicular magnetic fiel B . This will make electrons curve with a R.

We can relate the ions mass and the magnetic field considering two equations:

- Motion equation of ions

$$\frac{mv^2}{R} = q|\vec{v} \times \vec{B}| = qvB \quad (1.11)$$

- Conservation of energy

$$\frac{1}{2}mv^2 = qV \quad (1.12)$$

By extracting v^2 by one of the two and substitutin we get the following expression

$$\sqrt{\frac{m}{q}} = \frac{RB}{\sqrt{2V}} \quad (1.13)$$

1.2 CZ-growth

1.2.1 Doping concetration in the ingot

CZ growth of silicon gives different doping concentrations for the wafers depending on the distance from the top of the ingot.

The parameters that we need to have the doping concentration at a certain value x are the segregation coefficient k and the initial dose of doping C_0

$$C = C_0 k (1 - f)^{k-1} \quad (1.14)$$

where f is the % of the ingot where we are.

Dimostration	
Define	
<ul style="list-style-type: none"> • V_0, I_0, C_0 volume, concentration and number of impurities in the initial fused silicon • V_l, I_l, C_l same of before but in the melted silicon after a certain time t • V_s, I_s, C_s same of before but in the solid cristal 	
When a little volume of silicon solidifies the concentration of impurities in the melt varies as	
	$dI_l = -kC_l dV_s = -kC_l = \frac{I_l}{V_0 - V_s} dV_s \quad (1.15)$
so we can write	
	$\int_{I_0}^{I_l} \frac{dI_l}{I_l} = \int_0^{V_s} -k \frac{dV_s}{V_0 - V_s} \quad (1.16)$
doing the integration and getting rid of the log units we arrive at	
	$I_l = I_0 \left(1 - \frac{V_s}{V_0}\right)^k \quad (1.17)$
by using the following relation	
	$C_s = -\frac{\partial I_l}{\partial V_s} \quad (1.18)$
we get the result	
	$C = C_0 k (1 - f)^{k-1} \quad (1.19)$

Form the concentration we can derive the resistivity of the wafer as

$$\rho = \frac{1}{q\mu C} \quad (1.20)$$

1.2.2 Maximum pulling rate

$$v_p^{max} = \frac{1}{LN} \sqrt{\frac{2\sigma\epsilon k_m T_m^5}{3r}} \propto \sqrt{\frac{1}{r}} \quad (1.21)$$

1.3 Deal-Grove model

Using the correct tables parameters we can derive the coefficients B and B/A for wet or dry oxidation through their Arrhenius form

$$B = C_1 \cdot \exp\left(-\frac{E_1}{kT}\right) \quad B/A = C_2 \cdot \exp\left(-\frac{E_2}{kT}\right) \quad (1.22)$$

or in an alternative way with their expressions

$$B = \frac{2DHP_G}{N} \quad B/A = \frac{HP_G}{N\left(\frac{1}{k_s} + \frac{1}{h}\right)} \quad (1.23)$$

remember that

- B is related to the transport through the present oxide so it isn't dependent on crystal orientation.
- B/A is related to the interaction with the surface its activation energy it's $\simeq 2eV$ that is the energy to break one Si-Si bond and it's strongly dependent on crystal orientation in fact

$$\left(\frac{B}{A}\right)_{\langle 111 \rangle} = 1.68 \left(\frac{B}{A}\right)_{\langle 100 \rangle} \quad (1.24)$$

Notice that none of the mentioned parameters depends on pressure.

The model gives us the following expression

$$\frac{x^2}{B} + \frac{x}{B/A} = t + \tau \quad (1.25)$$

The boundary between the 2 regimes ,parabolic and linear, is given by the following thickness

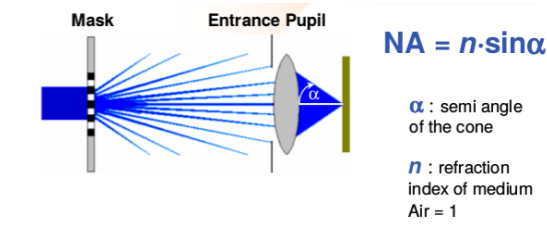
$$x_b = \frac{B}{2B/A} \quad (1.26)$$

Beware that 46% of the SiO_2 grows inside the silicon and the other 56% over it.

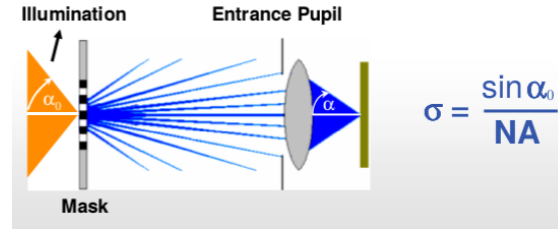
1.4 Lito

1.4.1 Resolution limit

Defining some parameters of our lito tool as:



NA determines the maximum number of diffraction orders that can be captured by projection lens and thus the quality of the reconstructed image.



It defines the illumination of the mask, depends from source extension. For $\sigma = 0$ we have coherent illumination $\sigma = 1$ incoherent illumination.

For a coherent illumination that means a light perpendicular to the mask we have that the minimum resolution is

$$R = \frac{1}{2} \frac{\lambda}{NA} \quad (1.27)$$

For a partially coherent illumination that means a light tilted by an angle θ the resolution is

$$R = \frac{1}{2} \frac{\lambda}{(1 + \sigma)NA} \quad (1.28)$$

Double patterning

With double patterning the R calculated before can be reduce further to $R/2$.

1.4.2 Contrast value

Given a certain profile of aerial image the contrast value is

$$C = \frac{I_{max} - I_{min}}{I_{max} + I_{min}} \quad (1.29)$$

With equal values of contrast it's better to choose the one with higher I_{max}

1.5 Technology scaling; propagation delay

Propagation delay of a metal wire can be modelled as

$$\tau \simeq 0.89 \cdot \varepsilon_{oxide} \varepsilon_0 \rho \frac{A}{F_{min}^2} \quad (1.30)$$

1.6 Electromigration

The critical length of the cluster can be expressed as

$$L_{crit} = \frac{2\sigma_{crit}\Omega}{Z^*q\rho J} \quad (1.31)$$

Where Ω is the volume of the atom of metal and σ_{crit} the value of the maximum stress.

1.7 Ion enhanced etch rate

Two process cooperating the ion etch and the byproducts deposition.

We can assume the number of deposited byproducts as

$$D = S_c(1 - \theta)F_c \quad (1.32)$$

With F_c flux of chemical species, S_c sticking coefficient and $(1-\theta)$ number of available sites. The number of byproducts removed as

$$R = K_i\theta F_i \quad (1.33)$$

with K_i sputtering efficiency, F_i flux of ionic species and θ number of occupied states.

In stationary conditions $D=R$ so we can extract θ as

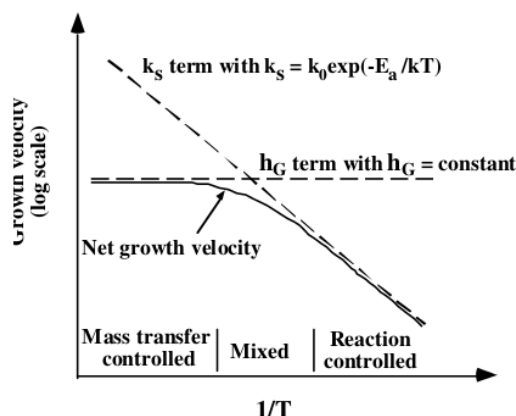
$$\theta = \frac{1}{1 + \frac{K_i F_i}{S_c F_c}} \quad (1.34)$$

so substituting this into the R expression we get the etch rate as

$$E = \frac{1}{N} \left(\frac{1}{K_i F_i} + \frac{1}{S_c F_c} \right) \quad (1.35)$$

with N density of the material to be etched.

1.8 CVD regimes



The velocity of deposition of a material in APCVD is determined by the slowest of two processes that operates in parallel; the transport of byproducts through the stagnant layer and the "reaction" with the surface.

The velocity can be mathematically described as

$$v = \frac{k_s h_g}{k_s + h_g} \frac{C_T}{N} Y \quad (1.36)$$

where k_s is the chemical surface reaction rate (has an Arrhenius form so strongly dependent on temperature) and h_g the mass transport coefficient (independent from temperature).

We can distinguish in two case

- $k_s \ll h_g$ we are in surface reaction controlled regime; the geometry of the reaction does not matter so much this is the best case
- $h_g \ll k_s$ mass transport controlled regime; the geometry of the reactor is fundamental to control the stagnant layer

We can increase the value of h_g since we hope to stay in surface reaction controlled regime decreasing the value of the pressure since

$$h_g = \frac{D_g}{\sigma_g} \quad (1.37)$$

where D_g is the diffusivity of the gas and σ_g the stagnant layer thickness and $D_g \simeq 1/P$.

1.9 APCVD rate

Dimostrazione

Only two fluxes are present in the CVD deposition

- F_1 flux of material near the surface.

It can be modelled as

$$F_1 = h_g(C_g - C_s) \quad (1.38)$$

where C_g is the concentration of species to be deposited in the bulk of the reactor, C_s the concentration near the surface and h_g is the mass transport coefficient.

- F_2 reaction and deposition flux at the surface.

It can be modelled as

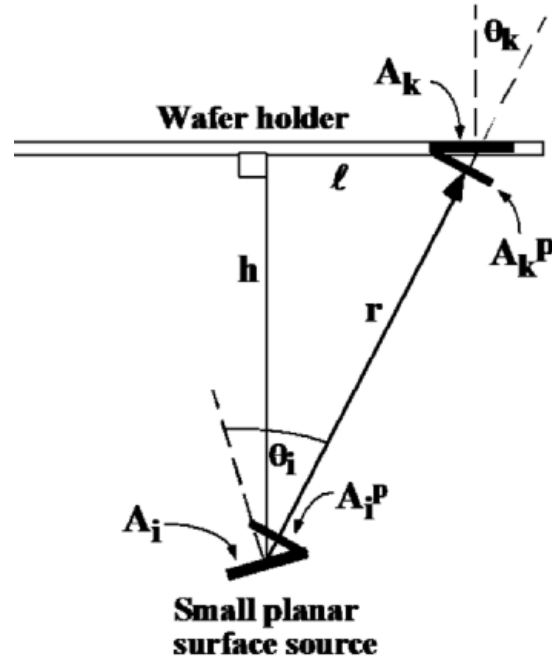
$$F_2 = k_s C_s \quad (1.39)$$

with k_s reaction rate at the surface.

In stationary condition this two have to be equal from this I can extract C_s and then remembering that $P_g/P_t = C_g/C_t$ we arrive at the following formula

$$v = \frac{F}{N} = \frac{k_s h_g}{k_s + h_g} \frac{P_g}{P_t} C_t \quad (1.40)$$

1.10 PVD evaporation from small source



The evaporation rate from a small surface is defined as

$$v = \frac{R_{evap}}{\pi N r^2} \cos(\theta_i) \cos(\theta_k) \quad (1.41)$$

where R_{evap} is the evaporation rate and N is the density of the material that we are depositing.