

UNIVERSITY NAME

DOCTORAL THESIS

Thesis Title

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*A thesis submitted in fulfillment of the requirements
for the degree of Doctor of Philosophy
in the*

Research Group Name
Department or School Name

September 26, 2016

Declaration of Authorship

I, John SMITH, declare that this thesis titled, "Thesis Title" and the work presented in it are my own. I confirm that:

- This work was done wholly or mainly while in candidature for a research degree at this University.
- Where any part of this thesis has previously been submitted for a degree or any other qualification at this University or any other institution, this has been clearly stated.
- Where I have consulted the published work of others, this is always clearly attributed.
- Where I have quoted from the work of others, the source is always given. With the exception of such quotations, this thesis is entirely my own work.
- I have acknowledged all main sources of help.
- Where the thesis is based on work done by myself jointly with others, I have made clear exactly what was done by others and what I have contributed myself.

Signed:

Date:

Chapter 1

Motivation and Background

1.1 Quantum info processing and Qubit candidates

1.2 Silicon vacancy as a Qubit candidate

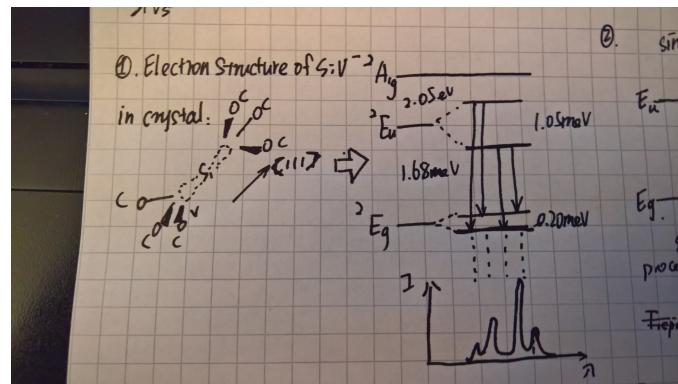


FIGURE 1.1

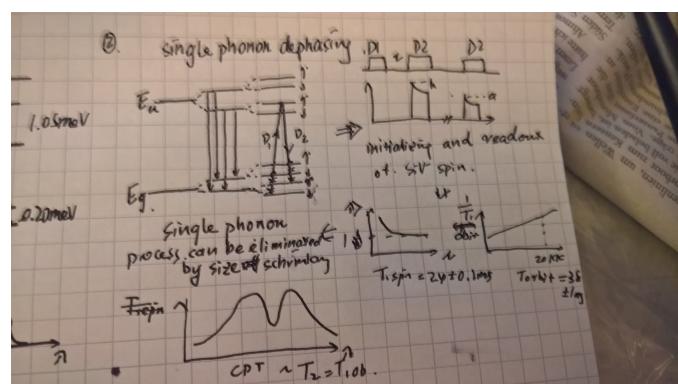


FIGURE 1.2

1.3 Silicon vacancies in nanodiamonds

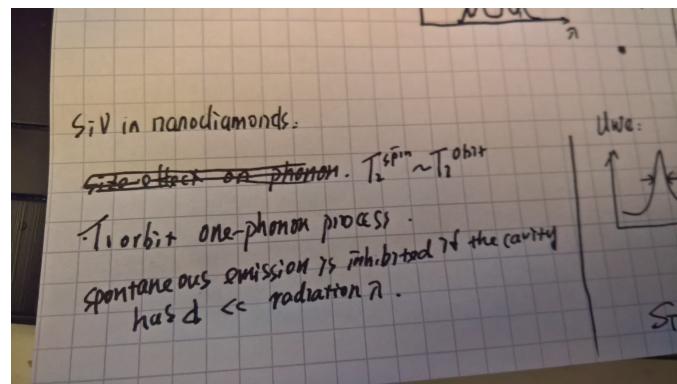


FIGURE 1.3

1.4 Motivation of the thesis, unsolved problem

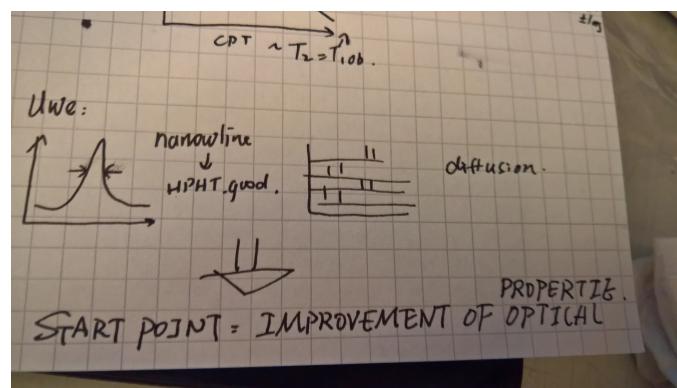


FIGURE 1.4

Chapter 2

Experimental approach of suppressing the spectral diffusion

2.1 Information regarding the nanodiamond sample

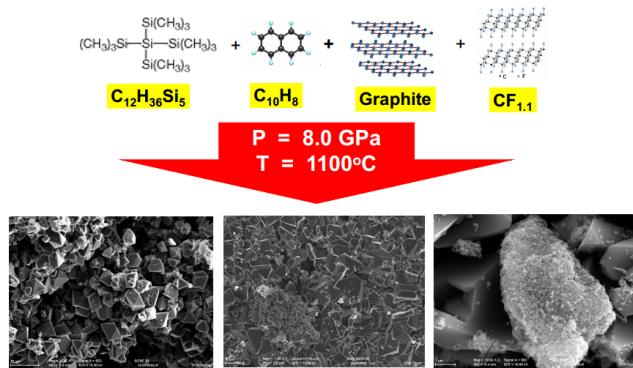


Fig. 1. SEM images of micro- and nanometer-sized Si-doped diamonds obtained as a result of high pressure high temperature treatment (HPHTT) of different mixtures of $\text{C}_{12}\text{H}_{36}\text{Si}_5$, C_{10}H_8 , $\text{CF}_{1.1}$ and graphite.

FIGURE 2.1

here is a paragraph about the fabrication of nanodiamond

After this the steps were the following:

1. Centrifuging at 2000rpm for 1 minute to remove the dirt. The liquid at the top was transferred to a new Eppendorf labeled "1st batch ND's". We suspect there to be not a lot of NDs as they are not soluble. These should be the smallest ones.
2. Diluting residues in Eppendorf with 1ml of microwater (the one behind the VWR) and then centrifuging at 1000rpm for 1 minute. The liquid part removed form this was named "2nd batch ND's"
3. Diluting residues again with 1ml of microwater. Centrifuging at 500 rpm for 1 minute. The liquid removed was called "3rd batch ND's".
4. Diluting residues again with 1ml of microwater. Centrifuging at 300 rpm for 1 minute. The liquid removed was called "4th batch ND's". This was the slowest possible centrifugation speed possible with this machine.
5. The residues after removing the liquid for batch 4 was called "residues after centrifuging at 300 rpm for 1 min"



FIGURE 2.2

here is a paragraph about the separation of the nanodiamonds

2.2 sample preparation

2.2.1 preparation of the substrate

IIa diamond as substrate To choose a proper substrate for the nanodiamond sample, a few principles need to be considered.

1. Low background fluorescence. It is always vital to obtain a decent signal to noise ration in any kind of measurements. As for our case, the emission(fluorescence) from silicon centers are the target, thus we would love to lower the back ground fluorescence as much as possilble.

2. Good heat conductivity at low temperature. From previous calculation done by Uwen Jantzen, we know that the temperature difference ΔT between the bottom of the substrate and nanodiamonds(which are spin coated on the surface of the substrate) can be estimate as

$$\Delta T = \frac{\sigma \cdot d \cdot T^4}{k}$$
, where σ is the Stefan–Boltzmann constant, d is the thickness of the substrate and k is the thermal conductivity. To resolve the fine feature of silicon vacancy ZPL, we want to characterise the nanodiamond sample at a temperature that is lower than 30K for spectrometer and 10K? for PLE.

3. No distracting spectral features. Some misleading peaks from the emission of the substrate would be the least wanted when we want to character a sample spectrally. In many cases, this is related to the raman-scattering of the photons, which highly depends on the crystal structure of the substrate. This scattering process alters the energy of the incident photons by shifts of concrete values and sometime can introduce peaks that are misleading or distracting.

4. Refractive index. Inam et al calculated the relative emission rate for radiating dipoles near an interface between two dielectrics with FDTD simulation. The result demonstrates that in both of the cases, when the dipole lies perpendicular and parallel to the substrate, the emission rate from a interface with lower relative refractive index is always higher than that from a interface with higher relative refractive index. And to increase the emission rate, a substrate with lower refractive index would be prefered.

Previously, taking these principles into consideration, my collegues have already ruled out a couple of materials, for instance, glass/quartz(distraction raman shift lines) and Sapphire(also a distracting raman shift line, and impurity induced emission that calls for extra attention when picking the optical filters). Now the temporary choice has landed on IIa type diamond, which has a low impurity density(resulting in low background fluorescence intensity), relatively low refractive index(2,4 to 2,7), good thermal conductivity($\Delta T = 4,17 \cdot 10^{-2} K$) and a raman shift at $1332 cm^{-1}$ that causes no distraction on our observation.

Focused Ion Beam milling In order to make it more convenient to trace the nanodiamonds, markers were curved onto the surface of the IIa type diamond substrate, this work was done by Uwe Jantzen during his master's

thesis period. As is shown in the fig.[], the focuses ion beam bombards the surface of diamond away and leaves behind markers that are visible in optical microscopy images and SEM images, as well as confocal microscopy images.

here is a sketch of how Ga-ion bombards the surface of substrate

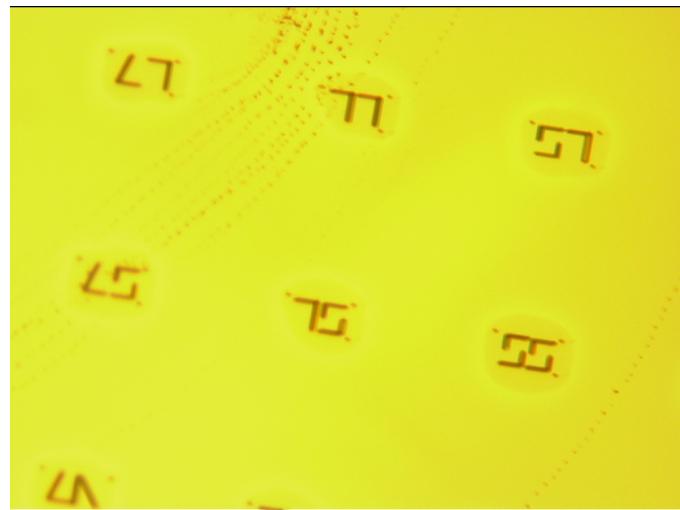


FIGURE 2.3

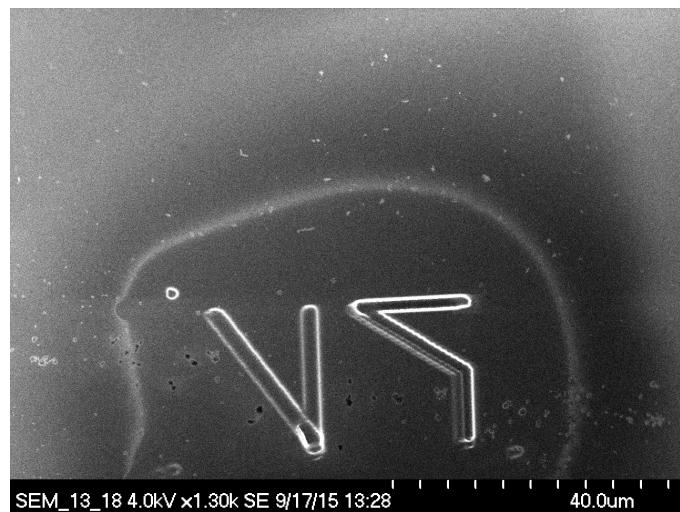


FIGURE 2.4

here insert image of markers, optical, sem and confocal

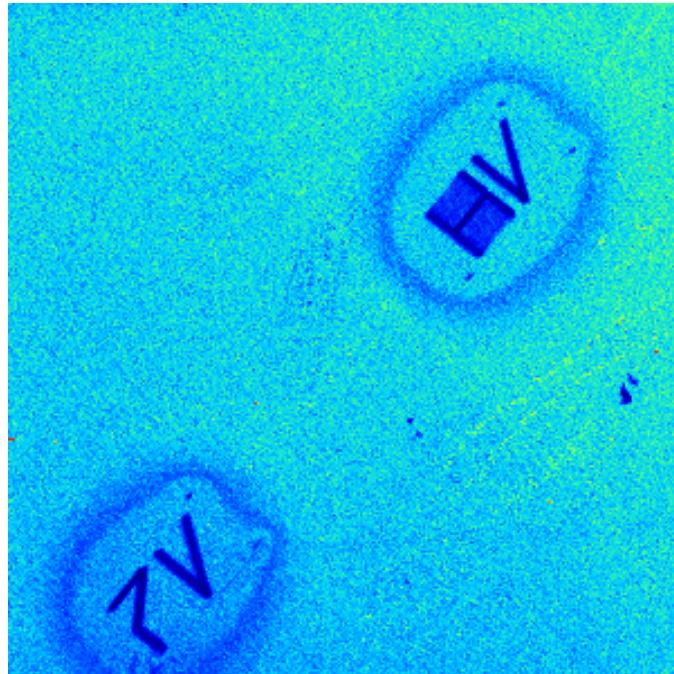


FIGURE 2.5

2.2.2 spin-coating of the sample

theory of spin coating

Spin coating is the method of sample preparing that mainly contains 2 steps:

1. Spreading of the liquid. In this step, certain volume of liquid containing the particle that we want to coat with is dropped on the surface of the substrate, driven by the centrifuging force from the rotational movement of the substrate, the liquid would be spread evenly on the surface.

2. Evaporation of the 'solvent'. While the sample stage rotates, the 'solvent' (In our case is not a real solvent, since nanodiamonds never really dissolve.) would evaporate, leaving the particle/molecules that are wanted to be coated on the substrate.

In this procedure, 2 factors we find important.

1. spin speed: generally the thickness of the liquid layer t is proportional to the inverse of the angular velocity w squared $t \sim \frac{1}{\omega^2}$, higher speed would help with forming a more uniform layer, yet this also means a smaller volume of solution, which would lead to lower density of nanodiamonds on the surface. On the other hand, with lower speed, the probability of aggregation would increase, which is also what we want to prevent.

2. volume of the 'solution': larger volume means longer drying time, which would increase the probability of aggregation and losing nanodiamonds, while smaller volume leads towards lower density of nanodiamond and more difficulty when trying to drop it with a pipette.

3. type of solvent: The type of solvent, viscosity and boiling point are important for the dispersion of nanoparticles inside solution, the spreading of the solution while spin coating and the rate of evaporation.

4.surface condition of the substrate. High contact angle is a obstacle towards the spreading of the solution, high roughness or inappropriate surface group of the substrate can result in poor wettability from the solution.

Throughout my project, with the help of Andrea Kurz, several combination of these factors had been has been tried out and in the end we landed on

here we need a list of different program of spin coating that we have tried,with indexxxxx

Acid cleaning

To make sure that the NDs dispersion can evenly spread and eventually settled on the substrate, a smooth, clean and hydrophilic surface is important.

Acid boiling is a very practicle way of diamond substrate cleaning. As it is called, the diamond will be boiled in a mixture of three strong mineral acids: sulfuric acid, nitric acid and perchloric acid. This mixture has ver strong ability of oxidizing.

here insert a sketch of how we do acid cleaning After assembling the setup, we initialize the reaction by heating the mixture to a temperature where is mildly bubbles. The substrate would be stay inside the boiling tri-acid mix for 4h. The mixture of strong mineral oxidizing acid can remove most of the adhesions on the surface of diamond substrate , leaving a clean hydrphilic surface. This oxidizing procedure will lead to the formation of carbonyl and carboxyl groups.

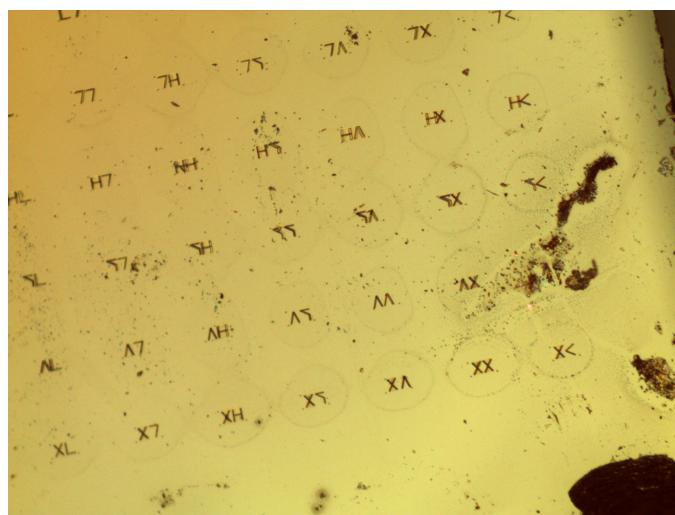


FIGURE 2.6

here insert image of before and after cleaning substrate, optical image, confocal image fig. the comparision between before and after acid boiling(need to be inserted later).

Tri Acid boiling blabla. Expectation of the surface. Before after cleaning. Optical image. Confocal image.

2.3 development of a technology to estimate the spectral diffusion

Setup In order to resolve the fine structure of ZPL of SiVs, we need to observe the sample at low temperature, thus a cryogenic setup must be applied. Our setup is a typical confocal microscopy setup connects with a cryostat, which cools the sample with liquid Helium flow.

here inserts a picture of our flow cryostat, from out and in side. This is the flow cryostat, whose main body is a vacuum chamber with

here insert a sketch of the cryo4 setup Confocal + Cryostat, Green laser + Red laser, spectrometer, apd, pic

PL Photoluminescence spectra is one of the most efficient way of finding silicon vacancies. In this measurement, we use green laser of 532nm to excite the Silicon vacancies from ground state to

PLE resonance excitation of optical transition. Resolution limited by scanning step of laser. Observing phonon side band with apd. range of scanning: limited by laser, small.

time resolved PL spectra Tracing PL spectra over time, show the diffusing behaviour of lines, characterisation methods: excitation polarisation: width of diffusion. Cross- correlation over time.

We recorded and noticed that the diffusion, whose range can up to 1nm, is far beyond the capability of PLE.

2.4 Oxidation

Effect of Oxidation Room temperature oxidization is a common way of nanodiamond purification. With different oxidizing temperature, different types of impurities can be removed from the surface of the nanodiamond, ranging from water and physisorbed organic impurities, amorphous carbon, and graphitic shells and ultimate the sp^3 phase of diamond[T.Gaebel,2010]. After the oxidation, carbonyl and carboxyl groups are formed on the surface[Petrakov,2012]. Several paper have mentioned temperature choices for oxidation aiming at impurity removal. During the master's thesis period, 2 different oxidation has been examined.

2.4.1 first Oxidation

method As reported, oxidation of sp^2 carbon already starts at $400^\circ C$, while the size reducing rate of diamond phase remains neglectable when the temperature is lower than $500^\circ C$. So it is an obvious choice to settle down the oxidation temperature at somewhere close to $500^\circ C$ when the maximum removal of sp^2 phased carbons and the minimum lost of the diamond

body(sp^3 phased carbons). Inspired by Elka Neu's paper,here insert a sentence explaining why we choose the two step oxidation program.

setup The aerobic oxidation is carried out in a tube furnace that is offered by the ?? institute and it is done with the help of Markus Mohr. The tube furnace consists of a glass tube connected to the room atmosphere and heating coils around the glass tube. The glass tube is slidable. We put our sample inside a ceramic ?bowl? and put the bowl into the glass tube carefully, after the temperature has been raised to ?460C?, the glass tube would be slide into the heating coils. After the Oxidation, the glass tube would be slide out and the sample would cooled inside the tube until room temperature.

pre-characterisation Before the oxidation, we tried to characterise the sample with several different methods based on our confocal microscopy setup.

1. RT poi mapping

First we take a scan with green laser and record the fluorescence with an APD, which offers us a confocal microscopy image of the surface of the sample, then the photoluminescence spectra of the bright spots are taken, those ones with a sharp peak at 737nm are saved as points of interests and their positions are saved as region of interest for the reference of further examine.

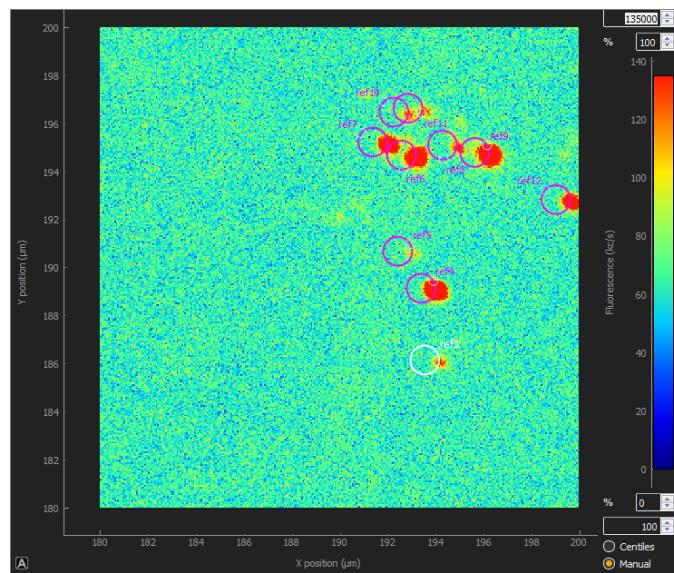


FIGURE 2.7

2. Cold spectra and PLE The sample is attached to an cold finger and the placed inside the cryostat, after UHV condition has been achieved, we start the helium transfer, which would brought the temperature of the sample down to 4.8K. We refound the points of interests that has been confirmed with SiV like spectra

3. PLE spectra

4.time-resolved PL spectra.

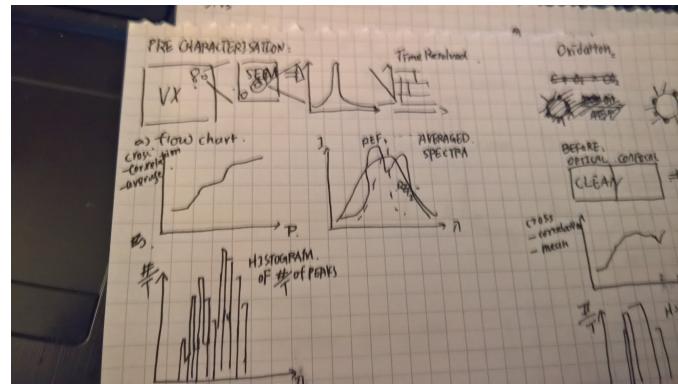


FIGURE 2.8

post-oxidation characterisation

1. Optical microscopy check The first thing we found after the oxidation is that, the surface of our sample turned very dirty. We are yet not certain about what the contaminations are, are they intrinsic or are they external. A possible deduction is that, the contamination comes from the glass tube of tube furnace, that the residues of previous treatments has attached to the inner surface of the tube and evaporized again, depositing on the surface of our sample. Further improvement of oxidation operation has been done in our second oxidation test, and will be mentioned in the next part of the thesis.

2. confocal microscopy imaging and PL spectra. Huge amount of bright spots can be seen in the confocal image when we excite the sample with 532nm green laser. There's no Silicon vacancy like spectra found in these bright spots. We refind our points of interests next to the marker 4C. The photoluminescence spectra shows much higher intensity than before the oxidation.

3. Cold spectra and PLE After the confirmation of points of interests, the sample was transferred into the flow cryostat and the helium flow brought the temperature down to 4.8K.

At 4.8K we recorded the time-resolved photoluminescence spectra of different incident beam power with an excitation wavelength of 532nm.

After the first oxidation, we learned that due to the inner strain of photonic fibre, the incident beam can not preserve a static polarisation. To stable the polarisation, we used a polarising beam splitter with a LC noise eater behind it. This would fix the polarisation at vertical direction.

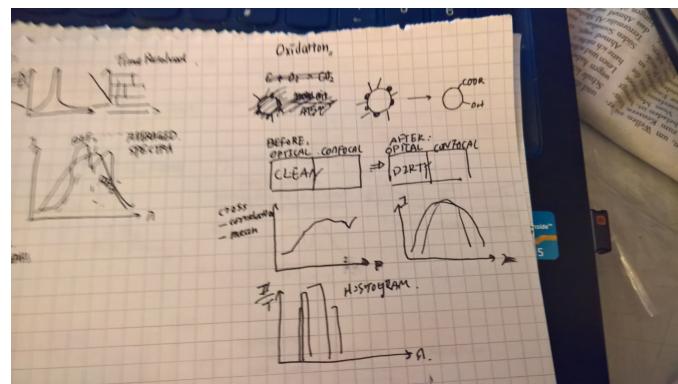


FIGURE 2.9

Analysis After the oxidation we noticed an increased intensity of emission, both background and the 737nm line. The number of lines per spectrum has decreased, while before the oxidation, fine lines ranging from

2.4.2 second Oxidation

method As is mentioned before, the optical properties of SiV in bulk diamond is extraordinary. Most importantly, the spectrodifusion that we have observed in nanodiamonds has never been seen in bulk diamonds. In the first oxidation, it seems the removal of graphitic impurity didn't help with the stabilization of emiision lines. In this second oxidation, we decided to used a higher temperature to acquire a surface with groups that imitates the bulk diamond. As reported by [paper], after 2 hours of aerobatic oxidation at 575°C , ... here insert a sentence of the surface groups of nanodiamonds. To increase the chance of finding smaller nanodiamonds that would fit into a cavity and decrease the chance of getting clusters of nanodiamonds, this time we chose to use nanodiamond of the first batch. These nanodiamond are spin coated on the substrate following the method II(the index, can be change).

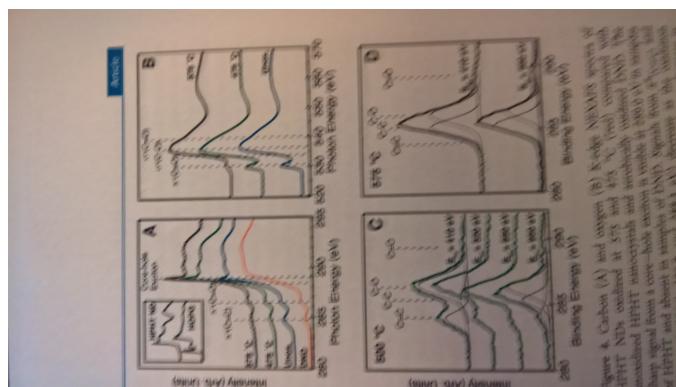


FIGURE 2.10

Taking the experiences of last oxidation into consideration. This time we introduces flowing inert gas (helium) to flush away the potential contaminations during the cooling process. This can also prevent the result to be affected by the humidity of the air. We found out the extinction rate of polarising beam spliter is not ideal, so this time we used a Clan Thompson polarisation filter instead.

Before Oxidation 1.Optical microscopy observation

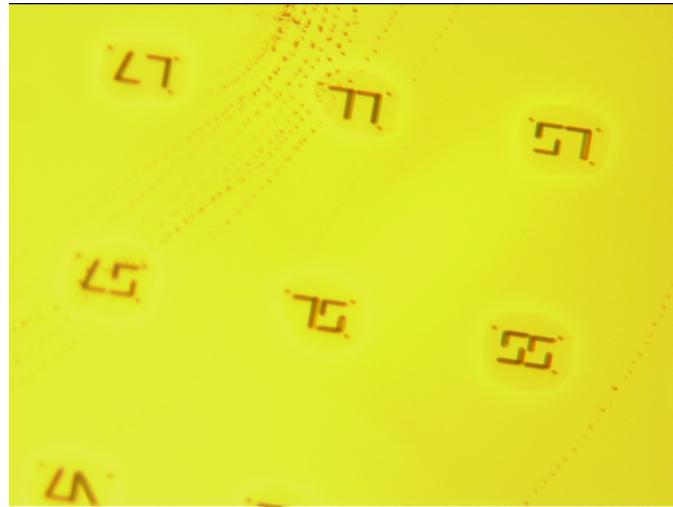


FIGURE 2.11

THIS IS NOT THE RIGHT FIGURE NEED CHECK THE SERVER We observed the sample after the spin coating with optical microscopy, the surface appeared to be relatively clean, little amount of contamination has been observed, but is acceptable.

2.Mapping of SiV^- with room temperature setup.

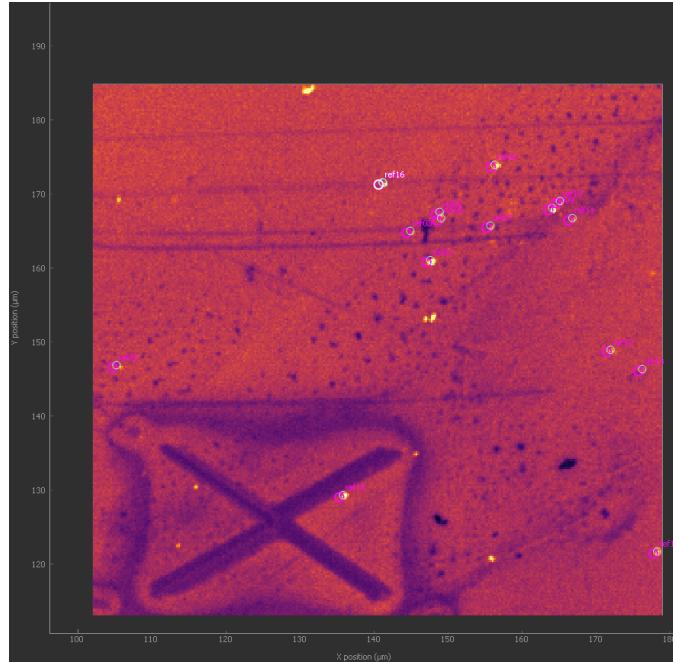


FIGURE 2.12

Once again, we explored the sample with the same room temperature confocal microscopy setup. With the help of spectrometer, we find a few points of interest with a emission spectrum that resembles SiV^- .

3. Cold time-resolved PL and exitation polarisation

As has been mentioned in last chapter, it is suspected that the incident polarisation can affect the spectral behaviour of SiV . We added in the excitation

polarisation measurement and recorded the time resolved photoluminescence spectra of 2 different excitation polarisation that are perpendicular to each other. This is achieved by putting a motor-driven half-lambda plate after the noise eater. Due to short time scheme from this measurement we decided to fix the input power at [?need to check], which is the lowest power that can offer most of the points of interest's a decent signal to noise ration.

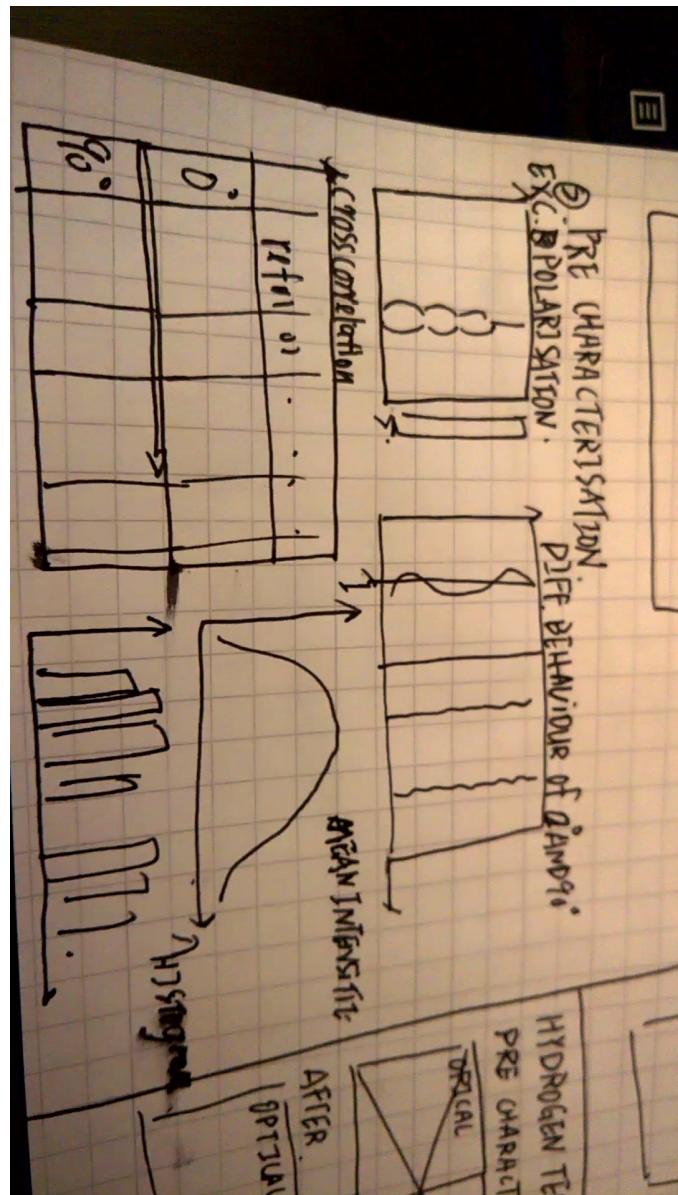


FIGURE 2.13

After Oxidation 1. Optical Microscopy observation.

After the Oxidation, we found the surface not as dirty as the last Oxidation. It seems a cleaner tube and flowing gas flushing do have helped suppressing the surface contamination introduced by the tube furnace. 2. Room temperature mapping

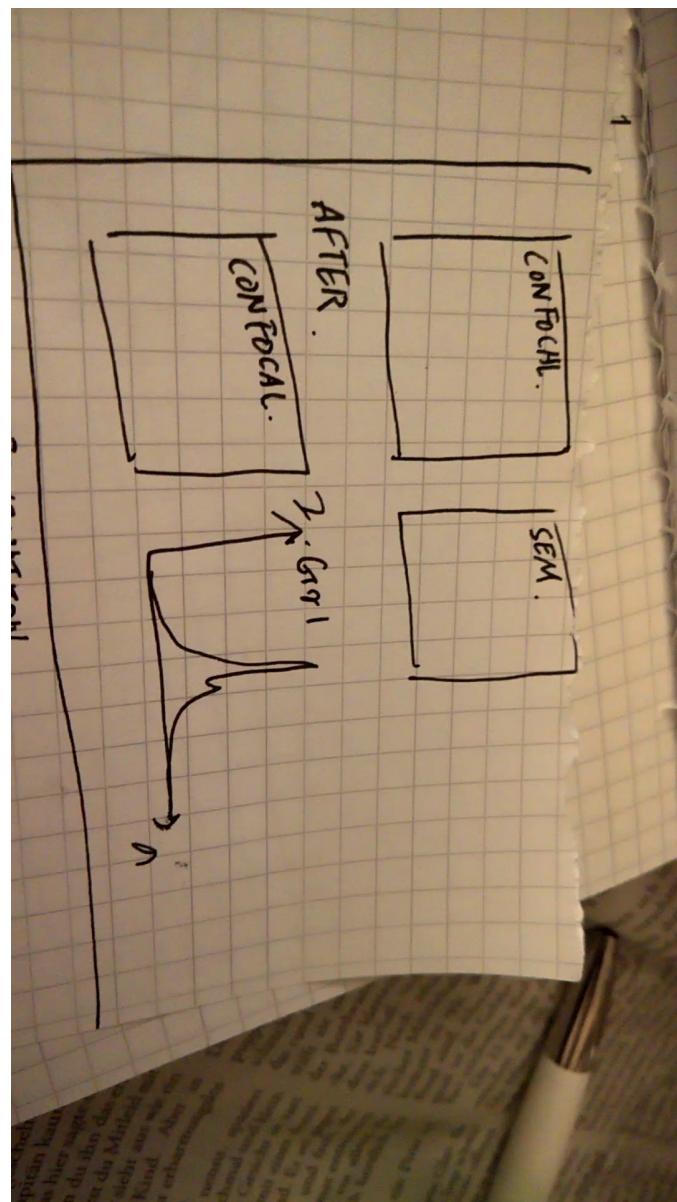


FIGURE 2.14

Analysis Comparasion if possible: different behaviour pre treatment between two batches Possible reason: losing NDs due to Helium flow while cooling, GR1 getting closer to the surface due to oxidation caused size/thickness reduction.

2.5 H termination

Effect of H termination NEA, band structure of diamond. Reduction of surface.

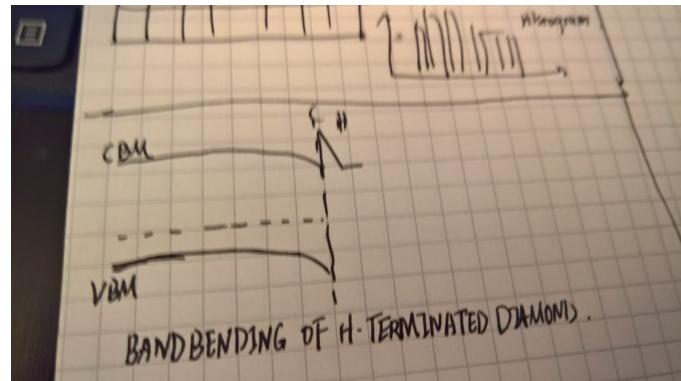


FIGURE 2.15

method Plasma treatment, setup, apparatus. ASK OSCHDI TO SEND THE PARAMETERS

why no pre characterisation Conditions for Plasma treatment. Vacum and clean surface.

After H termination Confocal image, optical image, excitation polarisation, time resolved PL with different incident polarisation.

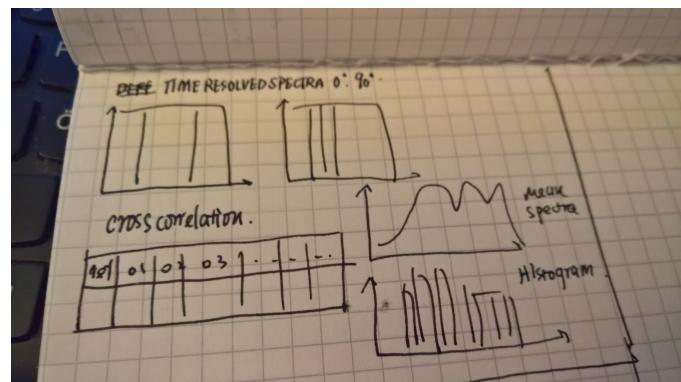


FIGURE 2.16

Analysis Within the instrumental limit of spectrometer, the spectral diffusion has been significantly suppressed. Possible reason.

Chapter 3

Conclusion and outlook

3.1 The road so far

Initial motivation

Development of a method to estimate the spectral diffusion

Surface treatments and their effects

3.2 Probabilities in the near future

PLE

life time measurement

comparasion of different surface group -OH and graphitic patch

better method for size selection porous/gel system, chromatographic way of size selection

relation between surface geometry and spectral behaviour

Appendix A

Appendix Title Here

Write your Appendix content here.

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