## Power-law creep and non-Newtonian viscosity

Newfouran fluiel: ==-p =+ C \for linear

Note: Constitutive law is linear but the

tin. momentum balance is still non-linear.

In Earth science most important non-Newtonian rheology is power-law creep:

¿ = strain rate

6 = stress

n = stress exponent

A = pre-factor



Note this is a scalar relation we med proper tensor form.

This behavior is common in polycrystalline solids close to their welting point.

We follow exposition in "Rheology of the Earth"

The general tensor form can be established from experiments.

Simple shear

$$d = \dot{\varepsilon} = \begin{bmatrix} 0 & \dot{\varepsilon}_s & 0 \\ \dot{\varepsilon}_s & 0 & 0 \\ 0 & 0 & 0 \end{bmatrix}$$

Uniaxial compression

$$d = \dot{\varepsilon} = \begin{bmatrix} \dot{\varepsilon}, & 0 & 0 \\ 0 & 0 & 0 \end{bmatrix}$$
 approx.  $\dot{\varepsilon}$ 

Suppose shear experiments lead to relation  $\dot{\varepsilon}_s = A \, \sigma_s^n$ 

A is function of P, T& material parameters but stress exponent n is constant.

How do we extend experimental result to

a general state of stress? What is tensor form of power-law errap?

- 1) Experiments are not affected by pressure  $\Rightarrow$  use deviatoric show & atrain rate

  2) Frame indifferent  $\Rightarrow$  use invariants

  Invariants from Lecture  $I_1(\S) = \text{tr}(\S) = \lambda_1 + \lambda_2 + \lambda_3 = S_{11} + S_{22} + S_{23}$   $I_2(\S) = \frac{1}{2} \left( \text{tr}(\S)^2 \text{tr}(\S^2) \right) = \lambda_1 \lambda_2 + \lambda_2 \lambda_3 + \lambda_1 \lambda_3$   $I_2(\S) = \left| \frac{3}{6} \right| \frac{3}{6} = \left| \frac{1}{6} \right| \frac{3}{6} = \frac$
- $I_{2}(\underline{S}) = -\left(\hat{\sigma}_{N} \hat{\sigma}_{22} + \hat{\sigma}_{N} \hat{\sigma}_{23} + \hat{\sigma}_{22} \hat{\sigma}_{23}\right) + \hat{\sigma}_{12}^{2} + \hat{\sigma}_{13}^{2} + \hat{\sigma}_{23}^{2}$   $I_{3}(\underline{S}) = \det\left(\underline{S}\right) = \lambda_{1} \lambda_{2} \lambda_{3}$

Not sure why  $I_3$  is not considered further likely  $I_3(\dot{\xi}) = \nabla_{20} \cdot \mathcal{V} = 0$ 

Invariants of deviatoric tensors o' è'

$$J_{1}(\underline{\varepsilon}) = J_{1}(\underline{\varepsilon}) = 0$$
 by definition

$$J_{2}(\underline{\delta}) = \underline{\lambda} \ \underline{\delta}' : \underline{\delta}' \qquad J_{3}(\underline{\dot{\epsilon}}) = \underline{\lambda} \ \underline{\dot{\epsilon}}' : \underline{\dot{\epsilon}}'$$

To see this

$$J_{2}(\underline{\sigma}) = \delta_{12}^{2} + \delta_{22}^{2} + \delta_{13}^{2} \quad \text{because } \delta_{11} = \delta_{22} = \delta_{23}^{2} = 0$$

$$\underline{\sigma}': \underline{\sigma}' = \delta_{12}^{2} + \delta_{21}^{2} + \delta_{13}^{2} + \delta_{31}^{2} + \delta_{23}^{2} + \delta_{32}^{2} = 2 \quad J_{2}(\underline{\sigma}) \quad \text{by symmetry}$$

If material is incompressible  $J_2(\underline{\dot{\epsilon}}) = I_2(\underline{\dot{\epsilon}}) = \frac{1}{2}\underline{\dot{\epsilon}} : \underline{\dot{\epsilon}}$ 

We can define effective stress and strain rate  $\delta_{\hat{E}} = \sqrt{\frac{1}{2}} \frac{\hat{E}}{\hat{E}} = \sqrt{\frac{1}{2}} \frac{\hat{E}}{\hat{E}} = \frac{1}{2} \frac{\hat{$ 

and rewrite the power-law as

$$\mathcal{E}_{E} = A \mathcal{E}^{n}$$

in terms of invariants and hence objective.

To extend this to tensorial form we assume

$$\underline{\dot{\varepsilon}} = \lambda(\delta_{\varepsilon}) \underline{\dot{\sigma}}'$$

which is reasonable for isotropic materials.

$$\dot{\mathcal{E}}_{E} = \int_{\overline{\mathcal{E}}}^{1} \dot{\mathcal{E}} : \dot{\mathcal{E}} = \int_{\overline{\mathcal{E}}}^{1} \lambda^{2} \dot{\mathcal{E}}' \cdot \dot{\mathcal{E}}' = \lambda \int_{\overline{\mathcal{E}}}^{1} \dot{\mathcal{E}}' \cdot \dot{\mathcal{E}}'$$

$$\Rightarrow \dot{\mathcal{E}}_{E} = \lambda \dot{\mathcal{E}}_{E}'$$

combining  $\dot{\varepsilon}_{\rm E} = A \, \delta_{\rm E}^{'n}$  and  $\dot{\varepsilon}_{\rm E} = \lambda \, \delta_{\rm E}^{'}$ we have  $\lambda \, \delta_{\rm E}^{'} = A \, \delta_{\rm E}^{'n} \implies \lambda = A \, \delta_{\rm E}^{'n-1}$ 

Tensor form of power-low creep  $\dot{\underline{\varepsilon}} = A \, \delta_{\varepsilon}^{\prime (n-1)} \, \underline{\sigma}^{\prime} \qquad .$ 

Compare to Representation Thu

$$\underline{\mathring{\mathcal{E}}}(\underline{\mathring{\mathcal{E}}}') = \alpha_{\mathbf{e}}(\mathbf{I}_{\mathbf{e}}) \underline{\mathbf{I}} + \alpha_{\mathbf{i}}(\mathbf{I}_{\mathbf{d}}) \underline{\mathring{\mathcal{E}}}' + \alpha_{\mathbf{z}}(\mathbf{I}_{\mathbf{d}}) \underline{\mathring{\mathcal{E}}}'^{2}$$
we see that  $\alpha_{\mathbf{e}} = \alpha_{\mathbf{z}} = 0$   $\alpha_{\mathbf{i}} = A \varepsilon_{\mathbf{e}}^{(n-1)} = \alpha_{\mathbf{i}}(\mathbf{I}_{\mathbf{i}}(\varepsilon'))$   $\checkmark$ 

⇒ frame indifferent

Example: Simple shear

Example: Dimple shew

$$\dot{g} = \begin{pmatrix} 0 & \dot{s}_{5} & 0 \\ 0 & 0 & 0 \end{pmatrix} = \underline{g}' \qquad \dot{g} = \begin{pmatrix} 0 & \dot{c}_{5} & 0 \\ \dot{c}_{6} & 0 & 0 \end{pmatrix} = \underline{g}'$$

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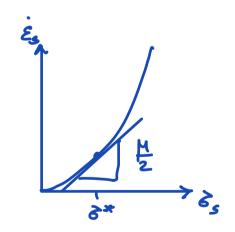
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Effective viscosity of power-law creep Standard Newtonian Fluid



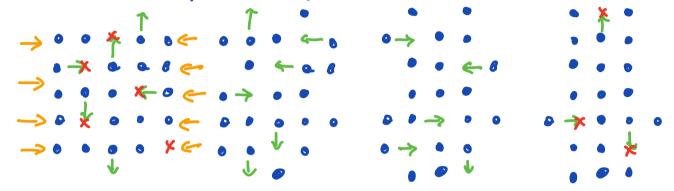
What are A and n?

⇒ depends on deformation mechanism

## Atomic basis of creep deformation Solid deforms by motion of lattice defects.

## I) Diffusion creep

Due to diffusion of lattice vacancies



Vacancies migrale to relieve the horizontal stress on the lattice. This process reduces the number of vacancies and hence the energy of the lattice.

Diffusion creep leads to Newtonian behaviour  $\stackrel{\sim}{\underline{\sigma}} = 2\mu \stackrel{\stackrel{\sim}{\underline{e}} \quad \text{with} \quad \mu = \frac{RT d^2}{z 4 V_a D_o} \exp\left(\frac{E_a + p V_a}{RT}\right)$ strongly grain size and temperature dependent.

## II) Dislocation Creep

Imperfections in crystal lattice

Edge dislocation

Skrew dislocation

