Data Driven Modeling, Statistical Analysis, and Machine Learning for Additive Manufacturing

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Abstract

In metal additive manufacturing (AM), materials and components are concurrently made in a single process as layers of metal are fabricated on top of each other in the (near) final topology required for the end-use product. Consequently, a large number of processing degrees of freedom (tens to hundreds) must be simultaneously controlled and understood; hence, metal AM is a highly interdisciplinary technology that requires synchronized consideration of physics, chemistry, materials science, physical metallurgy, computer science, electrical engineering, and mechanical engineering. The use of modern statistics-based approaches to modeling data sets with many degrees of freedom (known as machine learning) with metal AM can reduce the time and cost to elucidate and optimize the complex multidisciplinary phenomena. Machine learning techniques have been used in materials science for several decades. Most prolifically, the density functional theory community (DFT) rapidly adopted machine learning and has used it since the early 2000s for evaluating many combinations of elements and crystal structures to discover new materials. This focused review examines the potential of machine learning in metal AM, highlighting the many parallels to previous efforts in materials science and manufacturing, and discusses new challenges specific to metal AM.

CONTENTS

I.	Motivation	2
II.	Phrasing Additive Manufacturing as a Machine Learning Problem	5
	A. The Design Space of Additive Manufacturing	5
	B. Additive Manufacturing Data Types and Formats	6
	C. The Assumptions Behind Machine Learning	8
	D. Unsupervised Machine Learning	9
	E. Supervised Machine Learning	12
	F. Preparing data for machine learning	13
III.	Current ICME Tools are Well Equipped to Integrate with an ML Framework	13
	A. Pre-Build Design	13
	1. Alloy Design	13
	2. Design of Experiments	19
	3. Topology Optimization	23
	B. Process Design	24
	1. Model Complexity and Dimensionality Reduction	24
	2. Surrogate Modeling	28
	C. Process Monitoring and Characterization	32
	1. In Situ Process Monitoring and Feedback	33
	2. Featurization of Qualitative Image Data	35
IV.	Learning from the Past: Moving Towards Database-Driven Design of Additive	37
V.	Conclusions	41
	References	41

I. MOTIVATION

Metals-based additive manufacturing (AM) represents a paradigm shift in how products are manufactured, providing versatility in the type and design of parts produced by a single manufacturing facility, decentralizing manufacturing capability, and enabling novelty in the properties and design of the parts, to name but a few benefits AM offers. However, there have been significant roadblocks to fully realize AM's potential, particularly in the control of consistency and quality in part production and in the development of materials amenable to the AM process. Although decades of immense scientific and engineering work in industry, academia, and government have produced large advances in making AM a practical manufacturing solution, the metallurgical challenges facing AM persist. Computational materials simulation and the Integrated Computational Materials Engineering (ICME) approach have made strides in accelerating materials development. The ICME approach for AM incorporates investigations across many fields and scientific methods. The wide, diverse array of promises and problems in AM has resulted in a field of study which is rich with data. In this paper, we argue that data-driven, statistically-based machine learning can complement and accelerate the ICME process for additive manufacturing.

The 20th century saw the maturation of materials science and engineering as a field of study, enabling more rapid development of novel materials and materials manufacturing methods for specific applications. The process-structure-properties-performance paradigm transformed the combinatorial trial-and-error and intuition-driven materials discovery process into a problem of engineering a desired microstructure through designed processing. For instance, the history of turbine blade superalloy development is typified by advancements in control over microstructure through processing, including increasingly complex alloying recipes, multi-step heat treatments, and single-crystal casting.

In the past decades, materials development has greatly accelerated to match the broader acceleration of general technology advancement. Computational materials science has enabled the prediction of microstructure from processing and of properties from microstructure, reducing the need for costly and time consuming experimentation. The Integrated Computational Materials Engineering (ICME) approach tightly integrates physics-based computational models into the industrial design process, allowing the desired performance requirements of a part to guide the design of a novel material. Examples include low-RE Ni superalloys for better turbine performance and lower cost and the Ferrium S53 alloy designed for corrosion-resistant landing gears. Both cases took materials development timelines from decades to years, demonstrating the practical capability of designing new materials within an industrial product timeframe. Generalizing this capability to more industries and further

accelerating the process is the primary goal of the Materials Genome Initiative (MGI) [1].

Current AM capabilities are limited by materials-based problems which are uniquely difficult to solve with the above paradigm. The AM process itself is complex relative to traditional casting methods, including rapid solidification, vaporization and ingestion of volatile elements, and a complex thermal history, all of which vary with part location and require advanced computational tools to properly predict microstructure and properties. However, the lower cost and time barrier to entry for performing AM has enabled the rapid accumulation of experimental data, enabling the Edisonian approach for finding optimized AM processing methods and parameters of existing alloys. For AM, the ICME-based tools have been catching up, attempting to bootstrap legacy models to this data, with limited success. As such, current AM materials development is largely combinatorial, as the processing of legacy alloys are optimized with extensive design of experiments and AM-specific alloys with higher performance are just now being effectively developed.

It is within this context we argue that machine learning (ML) can accelerate the application of additive manufacturing. ML as a method for model development has shown wide application in the past years, in industries ranging from finance to social networking. The use of ML in materials has been relatively limited for a variety of reasons, primarily the lack of a curated and large dataset on which ML can operate. The MGI identified this as a primary problem for accelerating materials development, and there has been significant progress recently in infrastructure development for materials databases suitable for informatics tools such as ML. The difficulty in producing physics-based ICME models and the ability to rapidly and efficiently sample processing space in a combinatorial approach makes additive manufacturing an attractive use case for ML. Machine learning as a framework can couple the legacy ICME tools with the experimental data to produce much more accurate AM process-structure-property models and to automate the iteration of designed experiments for model improvement and optimized materials.

In this paper, we present our arguments for the use of ML to address AM challenges. We begin by phrasing terms and ideas from additive manufacturing in ways that are compatible with machine learning. We provide basics of machine learning algorithms and how they can be interpreted in an additive manufacturing study. Following this, we review machine learning methods that have been used in materials science previously. We attempt to state the uses of machine learning algorithms for addressing problems specific to additive. Follow-

ing this, we cover the use of machine learning in materials science previously. We conclude with an outlook for how database-driven design of materials for AM can accelerate material creation and part qualification.

II. PHRASING ADDITIVE MANUFACTURING AS A MACHINE LEARNING PROBLEM

Data-driven modeling and machine learning have been employed to great success across several fields including materials science. While machine learning may seem foreign at first, readers will see that it can be expressed and understood in plain terms. Many of the tenets and frameworks for machine learning are based in mathematical operations which will seem familiar to the reader, but applied in new ways.

A. The Design Space of Additive Manufacturing

The design space of additive manufacturing is the set of all machine parameters and measurable manufacturing phenomenon or material properties that result from the manufacturing process. An example of controllable machine parameters can be found in Table I. Measurable manufacturing phenomenon are quantities indicative of the processing history of the part. Examples include melt pool morphology, temperature, cooling rates, reflected intensity, and more. Measurable material properties are just that – aspects of the final part which are relevant to scientific or engineering applications.

The term design space will be used throughout this article to discuss the set of all additive manufacturing data which can be used with data-driven methods and machine learning algorithms. A single set of machine parameters, observed process phenomenon, and measured material properties can be considered as a *coordinate* in the design space. As an example, consider a list of parameters for a laser powder bed fusion process, such as [Alloy Composition, Laser Power, Scan Speed, Maximum Melt Pool Temperature, Average Cooling Rate, Final Density, Young's Modulus, Ultimate Tensile Strength]. Any part which is manufactured under a single set of conditions and is observed to have a set thermal history and material properties can be considered to be manufactured at that point in the design space.

TABLE I. A possible design space for laser powder bed fusion additive manufacturing. Any possible combination of these parameters is a point in the design space.

Parameter	range	step size	levels		
Skin/Contour/Core laser parameters:					
Power	$100-200 \ \mathrm{W}$	10 W	10		
Scan speed	$500\text{-}1000~\mathrm{mm/s}$	100 mm/s	5		
Spot size	50-100 $\mu\mathrm{m}$	$10~\mu\mathrm{m}$	5		
Energy density	$1-5 \mathrm{\ J/mm^2}$	$1~{\rm J/mm^2}$	5		
Build parameters:					
Polar angle	0-90°	30°	4		
Azimuth angle	0-180°	90°	3		
Sieve count	0-10	2	5		
Amount of recycled powder	0-100%	10%	10		
Other parameters:					
Blade direction	$0\text{-}300~\mathrm{mm}$	$10~\mathrm{mm}$	30		
Transverse direction	$0\text{-}300~\mathrm{mm}$	$10~\mathrm{mm}$	30		
Hatch spacing	$0.1\text{-}0.50~\mathrm{mm}$	0.1 nm	5		

Representing data in the design space properly will be a major pre-processing step for machine algorithms to be amenable to AM optimization.

B. Additive Manufacturing Data Types and Formats

Data types and sources in additive manufacturing are as widespread as any field of engineering and science. Machine learning algorithms, however, typically operate using specific mathematical representations of data. It is important to recognize all the different sources and formats of AM data and consider how they can be coerced into use for machine learning.

Scalar data values are some of the most common and easily obtained data in additive. A single set of manufacturing parameters, such as heat source parameters, can be represented as a list of scalars. Other scalar data can include material property measurements such as strength, hardness, density, moduli, and more. A good portion of this review covers how to model relationships between manufacturing parameters and material properties. In these

cases, the machine inputs will most often be represented as a vector, such as

$$\mathbf{x} = [\text{Machine Input 1, Machine Input 2, ...}]$$
..., Machine Input $n]$ (1)

out of n many scalar machine inputs. Vector representations of the design space are useful in determining manufacturing conditions which will result in similar results, as well as in building regression models of the AM process.

Another common data type is *time series* data. Time series data are usually collected from models of the AM process or in situ measurement. Probably the most common time series data collected in AM is temperature histories. These data can sometimes be used as-is, or operations can be performed to extract scalar data values from the time series. Scalar statistical values such as the maximum, minimum, mean, and standard deviation of a time series signal can be equally useful in machine learning applications.

Often, experimenters and modelers have quite a few data points already collected throughout the design space. They may have many vectors \mathbf{x}_i , each one containing machine inputs and measured or modeled material properties. It may make the most sense to represent the data as a matrix

$$\mathbf{X} = \begin{bmatrix} x_{1,1} & x_{1,2} & \dots & y_1 \\ x_{2,1} & x_{2,2} & \dots & y_2 \\ \vdots & \vdots & \vdots & \vdots \\ x_{n,1} & x_{n,2} & \dots & y_n \end{bmatrix}$$
 (2)

where the columns of X represent individual machine inputs or material properties and each row is a different measurement made somewhere in the design space.

Scalar, vector, and matrix representations of AM data will be among the most common types of inputs for machine learning algorithms. If a particular algorithm discussed differs from this paradigm, it will be explicitly noted.

A final concept which is core to machine learning is *covariance*. Covariance is measured between two data points $\kappa(\mathbf{x}\mathbf{x}')$, instead of being a property of a single data point. The covariance between data points encodes cross-correlated information within the design space. Ways of calculating covariance are many and varied and will be explicitly discussed where they are used. While many machine learning algorithms can operate directly on machine

inputs and processing outputs, it can be equally useful to calculate covariances between data points and use those as machine learning inputs.

C. The Assumptions Behind Machine Learning

Ultimately, AM users desire a model that relates manufacturing parameters to material properties; i.e., a process-property model. In a functional form, this model can be written as

$$f(x_1, x_2, ..., x_n) = \mathbf{y},\tag{3}$$

where the inputs x_i are n different manufacturing parameters and the outputs \mathbf{y} are measurable properties.

Currently, these process-property relationships are often developed by connecting separate process-structure and structure-property models of individual physical processes within AM. For example, individual process-structure models are developed to relate: heat source parameters to melt pool topologies [2] using the finite element method or similar computational techniques; melt pool topologies to solidification routes using thermal- and massdiffusion models [3]; solidification to microstructure evolution using phase field methods [4]. Separately, structure-property models are developed such as relating grain orientations and stress to material properties using crystal plasticity [5]. It is then by making connections between individual process-structure and structure-property models that researchers relate process parameters to material properties. This understanding-through-sequential-modeling approach has the added detriment that the errors and uncertainty of each approach compound on one another, so that the final result is less accurate, more time consuming, and more expensive than the direct build-break approach to develop process-property relationships. Machine learning is not a complete replacement for these traditional approaches, but rather a complementary modeling approach that can accelerate or even automate that process of building connections across many degrees of freedom that span the many different individual phenomenon within AM processes.

Two assumptions are necessary when using machine learning:

1. The Similarity Hypothesis: Parts manufactured at similar points in the design space will have similar properties.

2. The Relational Hypothesis: A correlative relationship exists between the data input to the model and the response of the system.

The first hypothesis is used to compare data and datasets, as well as to search and optimize through regression and classification algorithms. The second hypothesis is required for finding regression and classification functions that are physically accurate.

There are two types of machine learning covered in this review: unsupervised and supervised. Unsupervised learning will find trends in a dataset that are indicative of the underlying behavior. Supervised learning will learn a form of $f(\mathbf{x}) = y$.

D. Unsupervised Machine Learning

Unsupervised machine learning algorithms are used to identify similarities or draw conclusions from unlabeled data by relying on the similarity hypothesis.

Consider an experiment that varies three different manufacturing inputs x_1, x_2, x_3 and measures a single material property y. In matrix form, the data are expressed as:

$$\mathbf{X} = \begin{bmatrix} x_{1,1} & x_{2,1} & x_{3,1} \\ x_{1,2} & x_{2,2} & x_{3,2} \\ \vdots & \vdots & \vdots \\ x_{1,m} & x_{2,m} & x_{3,m} \end{bmatrix}$$

$$\mathbf{Y} = \begin{bmatrix} y_1 \\ y_2 \\ \vdots \\ y_m \end{bmatrix}$$
(4)

where $x_{i,j}$ is the j^{th} measurement of the i^{th} manufacturing input. A distance metric can be defined between data points in the design space. For example, data can be collected at two points $\mathbf{a} = (x_1, x_2, x_3)$ and $\mathbf{b} = (x_1 + \delta, x_2, x_3)$. The ℓ_2 norm of $\mathbf{a} - \mathbf{b}$ yields

$$||\mathbf{a} - \mathbf{b}||_2 = \delta. \tag{5}$$

The value and magnitude of δ gives an inclination about how similar **a** and **b** are. If δ is close to zero, then a researcher can say that they are similar, or even the same if δ is

exactly zero. As δ becomes larger a researcher can say **a** and **b** become more dissimilar. The concept of 'similar' manufacturing conditions may be easy to assess by an experimentalist when tuning only a few parameters at a time. When taking into consideration tens or hundreds of design criteria, sometimes with correlated inputs, elucidating similar manufacturing conditions becomes difficult. This vector distance approach is a simple, yet effective first glance at similarity in a design space and is generalizable to n many design criteria.

Let us say that δ is small and that \mathbf{a} and \mathbf{b} are similar manufacturing conditions. Now, consider a third point in the design space $\mathbf{c} = (x_1 + \delta, x_2 + \delta, x_3)$ that has not yet been measured. Since \mathbf{c} was manufactured at similar conditions to \mathbf{a} , as measured by $||\mathbf{c} - \mathbf{a}||_2 = 2\delta$, then we may say that \mathbf{a} , \mathbf{b} , and \mathbf{c} are all similar to each other. If the locality hypothesis is correct then manufacturing with conditions \mathbf{a} , \mathbf{b} and \mathbf{c} should yield similar measurements of y.

At some point, a researcher will have a set of initial manufacturing inputs **a**, **b**, **c**, **d**, etc., and associated property measurements that have been tested. Churning through the remainder of all possible manufacturing conditions becomes expensive and tedious quickly. Instead, researchers can use similarity metrics to determine whether or not a future test is worth running. Comparing the manufacturing inputs through vector distance gives a rough idea of the possible outcome before spending time and resources on running a test. If the intent is exploring design spaces then manufacturing at conditions *furthest away* from previously observed points may be the answer. If looking for local maxima of quality, an operator would want to manufacture at conditions *nearest to* the conditions currently known to have high quality.

Using vector distances as metrics of similarities can produce results that are analogous to creating process maps [6]. Process maps are used to divide 2 dimensional plots of manufacturing inputs into regions of quality, or regions of different material responses. The following demonstration is based on k-means clustering, a commonly used unsupervised machine learning clustering algorithm.

A researcher has acquired the datasets in Eqn. 4 and wants to partition \mathbf{Y} into groupings of high quality parts and low quality parts. However, there are several values of $y \in \mathbf{Y}$ that lie between two extremes and the cutoff for quality is not well defined. It would be useful to use similarity metrics to find the best possible partition of quality. To begin, the data set is partitioned randomly into two groups, \mathbf{Y}_1 and \mathbf{Y}_2 . The centroids m_1 , m_2 (or centers of

mass, in engineering) of each grouping can be calculated as

$$m_1 = \frac{1}{|\mathbf{Y}_1|} \sum_{y_j \in \mathbf{Y}_1} y_j$$

$$m_2 = \frac{1}{|\mathbf{Y}_2|} \sum_{y_j \in \mathbf{Y}_2} y_j.$$
(6)

where $|\mathbf{Y}|$ is the average value of a grouping. The measurements were randomly partitioned at first; the goal is to re-partition each set so that similar measurements (similar levels of quality) are in the same set. To do this, we can re-assign each set by

$$\mathbf{Y}_{1} = \{y_{i} : ||y_{i} - m_{1}||_{2} \le ||y_{i} - m_{2}||_{2}\}$$

$$\mathbf{Y}_{2} = \{y_{j} : ||y_{j} - m_{2}||_{2} \le ||y_{j} - m_{1}||_{2}\}.$$
(7)

We can interpret the re-assignment in Eqn. 7 physically: if a measurement initially assigned to set \mathbf{Y}_1 is closer in distance to \mathbf{Y}_2 then it is more similar to the other set. Thus, it is re-assigned. Since the original partition was random it is likely that there are low quality parts mixed in with high quality parts - in other words, outliers exist in each partition. Measuring the similarity of each data point to the mean of the groupings re-classifies these outliers into groupings that are more reflective of their quality.

Once re-assignment is complete the centroids in Eqn. 6 can be re-calculated and updated. Then, data points are re-assigned once more based on how similar they are to the centroid of each partition. If we have partitioned the input settings (x_1, x_2, x_3) along with their corresponding measurements, then we have lists of input settings which are likely to give good/bad quality parts. Further analysis can also be conducted, such as analyzing which regimes of inputs lead to good or bad quality - this is precisely what process maps represent. The difference in this case is that n many manufacturing conditions can be related to a quality metric simultaneously, with little to no human inspection or intervention. Additionally, a researcher can dig further and analyze why groups of input settings result in given quality for a material property.

E. Supervised Machine Learning

In a supervised machine learning algorithm the goal is to find a functional relationship that best approximates the underlying physical relationship $f(\mathbf{x}) = \mathbf{y}$. That is, supervised machine learning algorithms relate model inputs to labeled data. Functional relationships can take many forms, depending on the specific supervised ML algorithm being used. One method is to model the relationships as a vector product

$$XT = Y. (8)$$

where T is a vector of coefficients that weigh the machine inputs to approximate an entry in \mathbf{Y} .

A researcher usually seeks this relationship through the measurements they have observed; in this case, the measurements are stored in the matrices of Eqn. 4. A common method to find a vector representation of T, and a critical element in most machine learning algorithms, is through least squares regression. Least squares regression finds T through a minimization problem, given by

$$\min ||\mathbf{X}T - \mathbf{Y}||_2^2. \tag{9}$$

Equation 9 can be interpreted analogously to similarity measurements for unsupervised algorithms: the closer that $\mathbf{X}T - \mathbf{Y}$ is to zero, the more similar T is to $f(\mathbf{x})$.

The methods of solving equation 9 are many and varied; indeed, much of this review will focus on finding solutions to Eqn. 9 for various problems throughout additive manufacturing. The result is an approximation to the functional relationship $f(\mathbf{x}) = \mathbf{y}$. A new point of interest in the design space \mathbf{x}' can be chosen and its associated material property \mathbf{y}' can be predicted by computing

$$\mathbf{x}'T = \mathbf{y}'. \tag{10}$$

This simple example demonstrates how functional relationships can elucidate more information about design spaces. Commonly used machine learning algorithms in materials science and engineering are given in Table II. Note that the field of Machine Learning is evolving as fast as AM itself; hence Table II is by no means a comprehensive review of all machine learning algorithms, but rather is intended to provide a comparison between the

form and function of some of the most widely adopted algorithms in materials science and engineering.

F. Preparing data for machine learning

Table XX highlights a variety of computational tools and Python packages and their relevance to AM synthesis optimization.

III. CURRENT ICME TOOLS ARE WELL EQUIPPED TO INTEGRATE WITH AN ML FRAMEWORK

The following section details how machine learning approaches can tie-in to current efforts in AM. Since the study of AM is often focused around specific problems, this section is tailored to many common areas of study within AM. Different methods of analysis and characterization for different aspects of AM all tie-in well with different machine learning algorithms. Even more so, ML can be used to automate the generation of knowledge about AM process-structure and process-property relationships.

This article is not an exhaustive review of machine learning algorithms. The algorithms discussed were chosen because they were previously demonstrated in a materials science and engineering applications **or** because the possible application of an algorithm to AM was clear and immediate. Similarly, the additive manufacturing problems addressed are not all-encompassing. The AM problems discussed are those which the authors felt most immediately addressable with machine learning.

A. Pre-Build Design

1. Alloy Design

Should find a place to fit in a citation of Liu2006 [12] – it is a top-to-bottom ICME design study focused around mining on compositional information – very relevant to this section. They apply partial least squares as their data-driven method. Choice of alloy impacts the physics of AM from start to finish, starting with the optics of energy sources incident on feedstock and ending with the material properties of

TABLE II. Several of the most widely used machine learning algorithms that have been used in materials science are compared.

Class of	Examples	Applications	Strengths	Constraints
Algorithm				
Weighted	Decision	Regression,	These algorithms are	They can be suscepti-
neigh-	trees, k-	Classification,	robust against uncer-	ble to classification bias
borhood	Nearest	Clustering and	tainty in data sets and	toward descriptors with
clustering	neighbor	similarity	can provide intuitive re-	more data entries than
			lationships between in-	others.
			puts and outputs. See	
			Ref. [7] for a primer on	
			clustering.	
Nonlinear di-	t-SNE,	Dimensionality	These algorithms are	Projections can rep-
mensionality	Kernel ridge	reduction,	robust against	resent unphysical,
reduction	regression,	Clustering and	nonlinear input/output	difficult to interpret
	Multidimen-	similarity,	relationships and can	relationships. Global
	sional metric	Input/output	provide intuitive	relationships can also be
	scaling	visualization,	projections of the	lost when nonlinear di-
		Descriptor	material input/output	mensionality reduction
		analysis,	space. For accessible	results are projected
		Regression,	examples, see Refs.	onto lower-dimensional
		Predictive	[8, 9].	spaces.
		modeling		
Linear di-	Principle	Dimensionality	This type of algorithm	The relationships stud-
mensionality	component	reduction, Clus-	can produce orthogonal	ied must be linear in
reduction	analysis	tering and	basis sets which	nature, and these algo-
	(PCA),	similarity, In-	reproduce the training	rithms are susceptible
	Support	put/output	data space. They can	to bias when descriptors
	vector	visualization, De-	also provide quick and	are scaled differently.
	regression	scriptor analysis,	accurate regression	
	(SVR)	Regression, Pre-	analysis. For a primer	
		dictive modeling $_1$	on PCA specifically, see	
			Ref. [10].	

TABLE III. Common tools and packages for preparing data for ML applications

Data source (instrument/software)	file type	processing tools	"ML-ready
X-Ray tomography (Zeiss Xradia Versa)	.tiff	image featurizer	vectorized
X-Ray tomography (TRACR)	$.\mathrm{txt},.\mathrm{csv}$	Python (pypif, pandas, citriantion-client)	porosity st
X-Ray diffraction (Panalytical empyrean)	.xy	Python (pypif, csv, pandas)	XRD feat
Optical microscopy (Keyence VHX-5000)	.tiff	image featurizer	vectorized
Uniaxial tensile testing (MTS 370)	.xy	Python (Hough transform algorithm)	real-value
Composition (TOF-SIMS)		Python (pymatgen, matminer, magpie)	composition
Semi-structured web page	.html	Python (requests, beautiful soup)	csv-format
Semi-structured pdf (txt-based)	.pdf	tabula, Abbyy Fine Reader, Python (pdftotext) $$	csv-format
ML packages		Python (sklearn, tensorflow, keras, OpenCV)	

the final part. The reflected/absorbed intensity of lasers on powder beds is determined by the powder's composition [13, 14]. The density of powders, both intra- and inter-granular density, plays a role in final part density [15]. Conduction modes in the melt are partially determined by the thermal properties of the alloy [16]. All of this is not to mention that different alloys exhibit different grain morphologies and therefore microstructures [17].

Problems in the additive process can also be linked to composition such as vaporization of constituent elements due to rapid thermal fluxes, impacting the stoichiometry of melt pools and, ultimately, quality [18]. Even traditional engineering alloys sometimes need to be altered to improve compatibility with AM. Searching for new alloys specifically for AM may also be fruitful, as unique strengthening mechanisms can arise [18, 19]. Designing alloys for AM – either altering currently used alloys or starting from scratch – requires taking into consideration the compatibility of alloys' physical properties with AM. Alloy design for AM must take into consideration general alloy properties, like melting point, to feedstock-level properties, like vaporization temperature, to bulk alloy properties, like strength. All of this information has been documented and exists in databases that can be searched. Alloy designers from AM should take advantage of these databases to search for new alloys for use in additive.

Databases exist which contain alloy properties ranging from the reflectivity of the alloy to the mechanical properties of alloys in bulk. The International Crystal Structure Database (ICSD) contains crystallographic information for millions of compositions. The Linus Pauling files contains a range of material information, from atomic properties like radius and electron valency to crystallographic level information [20]. In modern day, large databases such as AFLOWLib [21], the Materials Project [22] and more allow users to search through large databases of relevant alloy information to find one that matches a desired property. Searching through large databases of information to find optimal compositions for manufacturing is actually one of the earliest materials informatics problems ever addressed. Methods exist to perform these searches in a fast, automated way. These methods are referred to as database mining, a data-driven materials design approach.

A study by Martin et al. used database mining to find micronucleants for Al alloys in powder bed manufacturing [16]. Part of the design process is identifying which alloy properties are important for the desired application. Heterogeneous nucleation of Al grains was the desired outcome of Martin's study. To induce such nucleation, Martin et al. searched for possible nucleants whose crystallographic lattice parameters closely matched that of Al. This way, the Al grains would have a low-energy-barrier nucleating site from which to grow heterogeneously. Martin's study employed a search algorithm to search through 4,500 different possible nucleants and identify those with the closest-matching parameters. Ultimately, Zr was found to be the best candidate.

The same process employed by Martin – identify the properties which need to be satisfied, then search for a material that is closest matching – can be extended to many other alloy properties relevant to AM. Database mining was first introduced in material science to predict stable compositions, or estimate material properties from composition. Database mining has been successfully implemented to predict stable crystal structures [23–25] and predict material properties as a function of composition [26–30]. Some specially designed search algorithms have also been designed for improved speed in automated searches [31]. Successes have been found in designing Heusler compounds using high throughput search methods [32]. Reviews of early high-throughput searches for compositions with ideal properties can be found in [33, 34]. The same search algorithms employed in these studies can be extended to the additive case.

A limiting factor in database mining is that designers are limited to properties which have been measured or calculated. We do not have information about the vast space of *possible* materials. Consider a set of alloying elements for Ti such as {Al, V, Zr, Cr}. Researchers may need to test the impact of alloying composition on the dendrite arm spacing of Ti alloys. Phase field models exist which simulate the growth of and measure dendrite arm

spacing as a function of a continuum of composition. A particularly efficient combined phase field/cellular automata model was implemented by Tan et al. precisely to model dendrite arm spacing in laser manufactured alloys [3].

Modeling all possible combinations of {Ti, Al, V, Zr, Cr} is possible with coarse additions of alloying elements, but undesirable. Machine learning can aid in the process to find an optimal composition without modeling all possibilities. *Genetic alogrithms* (GA) can search the space of possible alloys to find the optimal dendrite arm spacing. Genetic algorithms have been one of the most-used data driven approaches in materials science over the past few decades [25, 31, 35–39].

The principle of genetic algorithms is to evaluate the *fitness* of a population of candidate alloys against a *fitness function*. The fitness function is a method of evaluating how well a candidate alloy meets a criteria.

As a thought experiment, consider using Tan's phase field/cellular automata model as a fitness function. It can be run for candidate compositions – in this case, various amounts of {Al, V, Zr, Cr} alloyed into Ti – and used to evaluate the dendrite arm spacing. Once a fitness function has been identified, the next step in a genetic algorithm is to represent candidate alloys as a *gene*.

We can represent a gene as

Alloy =
$$[\chi_1, \chi_2, ..., \chi_n]$$

where χ_1 is the species and weight percent of the first element (titanium, in this example), χ_2 is the species and weight percent of the second element, up to n elements. For example, Ti-6Al-4V would be represented as

The goal is to find the alloy with optimal dendrite arm spacing. First, a population of candidate genes needs to be generated, either randomly or by design. Two examples from a starting population may be

Alloy 1 =
$$[0.9 \text{ Ti}, 0.05 \text{ Al}, 0.05 \text{ V}]$$

Alloy 2 = $[0.9 \text{ Ti}, 0.1 \text{ Zr}]$

The dendrite arm spacing can be estimated using Tan's model. It is not guaranteed that the optimal composition is contained in this starting population.

Genetic algorithms select genes out of the current population – called the parent generation – to proceed to another generation of model assessment – called the child generation. Selection consists of keeping the best performing compositions, say the top 10%, and discarding the rest. Genetic algorithms find optimal locations in the design space by relying on the similarity hypothesis. If one alloy is in the top 10% of genes then it is possible that a similar alloy will also be high performing – it may be even perform better. Once selection is done, the next step is to search the space near the best performing alloys from the parent generation.

Genetic algorithms generate similar compositions from those selected in the parent generation by making alterations to genes. One operation is *mutation*, whereby entires of the genes are changed. For example, we could mutate alloy 1 by changing the composition:

Parent Generation: Alloy
$$1 = [0.9 \text{ Ti}, 0.05 \text{ Al}, 0.05 \text{ V}]$$

Child Generation: Alloy $1 = [0.9 \text{ Ti}, 0.02 \text{ Al}, 0.08 \text{ V}]$

where in the child generation the amount of V was increased, while the amount of Al was decreased. Another operation which may be performed is *crossover* where entries of genes are added or interchanged. For example, one crossover operation may look like

Parent Generation: Alloy 1 =
$$[0.9 \text{ Ti}, 0.05 \text{ Al}, 0.05 \text{ V}]$$

Alloy 2 = $[0.9 \text{ Ti}, 0.1 \text{ Zr}]$

Child Generation: Alloy 1 = $[0.9 \text{ Ti}, 0.05 \text{ Al}, 0.05 \text{ Zr}]$

Alloy 2 = $[0.9 \text{ Ti}, 0.1 \text{ V}]$

where in the second generation V and Zr have been interchanged.

Selection, mutation, and crossover followed by model assessment and further selection, mutation, and crossover continues until the design criteria is met. The intuition behind genetic algorithms is that eventually the selection process is narrowed down to alloys within a given region such that further mutation and crossover do not produce new genes. Eventually, all the 'fittest' genes as determined by the model will converge to be approximately the same

composition.

Genetic algorithms have been applied to alloy design for low and high temperature structural materials [26, 40], ultra high strength steels [41], specific electronic band gaps [42], minimum defect structures [43], exploring stable ternary or higher alloys alloys [37, 44], and more. For a review on the application of GA's to alloy design through the early 2000s see Ref. [45].

Other machine learning algorithms have also been applied for classification and optimization of alloy compositions. Anijdan used a combined genetic algorithm—neural network method to find Al-Si compositions of minimum porosity [43]. Decisions trees have been implemented for a number of different alloy optimization, such as predicting ferromagnetism [46] and the stability of Heusler compounds [47].

In the search for new alloys, some compositions definitely won't be compatible with the additive process. It would be useful to identify these alloys up front or as quickly as possible. Additive can also be improved by machine learning algorithms which suggest the best materials or properties to test. A wide range of machine learning algorithms can be implemented to guide the entire experimental design process so that an optimized property is found as quickly as possible.

2. Design of Experiments

Parametric analysis, broadly defined, is an experimental method of mapping independent variables to their corresponding dependent parameters. Process-property relationships are typically studied through parametric analysis. Machine learning aids in investigations of AM by reducing the amount of experiments needed to characterize process-property relationships. Machine learning approaches like sequential learning model relationships in parametric studies to discover regions of the parameter space which will produce the most information about process-property relationships.

In additive manufacturing independent parameters such as laser energy, speed, build direction, composition, layer height, and more are varied to study their impact on material properties. Examples include relating build geometry to microstructure or surface roughness [48, 49] or temperature history to microstructure [50, 51], or substrate temperature to residual stress development [18, 52], or even entire manufacturing processes to microstructure

ture [53]. One of the most common types of parametric analysis is relating heat source parameters to all aspects of AM, such as part temperature history [54, 55], microstructure [56, 57], mechanical properties [58, 59], residual stresses [60, 61], and more.

Both engineering and scientific investigations of AM utilize parametric analysis. In the sciences, parametric analysis proceeds until a theory or model can be presented for a process-property relationship. In engineering, parametric analysis continues until an optimality criterion is met, such as maximum strength or minimum porosity. Both disciplines vary independent parameters and measure dependent responses; doing so provides information about the underlying phenomenon.

Information is any observation of process-property relationships. For example, observing that a set of laser parameters results in an equiaxed microstructure can be considered information because the researcher has gained an idea of the properties to expect from set processing conditions. Therefore, information gain is any experiment which reveals a previously unobserved process-property relationship. Rigorous mathematical definitions of information and information gain have been defined, typically referencing back to Shannon's original formulation of information theory [62].

Traditional design of experiments maximize information gain by dividing up the design space to maximize distance between like experimental conditions. Machine learning driven design of experiments makes suggestions for future experiments based on the results of past experiments. It does not make any assumptions up front about correlations between design parameters. Rather, machine learning algorithms model relationships between inputs and outputs and suggest the experiment which is statistically most likely to result in information gain. Design of experiments with machine learning algorithms can be adopted by augmenting traditional design of experiments with a statistical model.

The first step is to identify which parameters which may impact properties, as is done in traditional design of experiments. As more parameters and finer step sizes are added, the size of the design space grows. Once the scope of the design space has been defined, the next step is to generate an initial dataset. The process-property relationships revealed in these initial tests will be the basis of an initial statistical model.

After an initial dataset is generated, the researchers need to define a response function which interprets the relationship between parameters and material properties. One example is a regression model of the process parameters and material characteristics. Identifying

tests of importance begins by identifying all n parameters in the design space that may or may not impact a final manufacturing result y. First, a subset of the n many parameters m_1 is chosen such that at least one parameter in \mathbf{x} is left out. A simple regression function called a decision tree is trained

$$\hat{f}_1(\mathbf{x}_{m_1}) = y \tag{11}$$

on the subset of features m_1 . Next, a new random sampling of the n parameters m_2 is chosen, and a second regression problem is solved $\hat{f}_2(\mathbf{x}_{m_2}) = y$. A random forest is a linear combination of all of these individual regression functions

$$\hat{f}(\mathbf{x}) = \frac{1}{B} \sum_{b=1}^{B} \hat{f}_b(\mathbf{x}_{m_b})$$
(12)

out of B many parameter samplings. A standard deviation in the prediction output is typically computed as

$$\sigma = \sqrt{\frac{\sum_{b=1}^{B} \hat{f}_b(\mathbf{x}_{m_b}) - \hat{f}(\mathbf{x})}{B - 1}}$$
(13)

Some of the features in \mathbf{x} may be irrelevant for studying process-property relationships and some may be very important to model. Random forests test the importance of an AM process parameter in \mathbf{x} by re-training every individual regression function $\hat{f}_b(\mathbf{x}_{m_b})$ with that machine input included. Then, the standard deviation is re-computed. If the inclusion of a parameter in every regression tree significantly changes σ then it plays a large role in prediction of y. Therefore, the feature should be included and tested in simulations.

Ling et al. performed design of experiments to optimize the process of designing high fatigue strength steels. [63]. In Ling's case, a random forest model was fit to design criteria from a database of steel compositions, processing routes, and fatigue strength. The goal was to find the composition and processing route combination which had the highest fatigue strength in the dataset using as few experiments as possible. A benefit of random forest models is that regions of the design space with high uncertainty can be identified.

Random forests are easy to use because they do not require tuning many hyperparameters. Furthermore, random forests are very good at sifting through many irrelevant features to find the features that actually matter. Random forest training is also computationally inexpensive and easily parallelized. Random forests provide two advantages that are particularly important in the context of materials science: the ability to efficiently calculate uncertainty estimates and the added interpretability of feature importance metrics. Based on their ensemble nature, it is possible to generate uncertainty estimates for random forest predictions using jackknife-based procedures [64–66]. They automatically identify the most important features in a given model based on how often a feature is used in splitting criteria and what the aggregated information gain is over those splits. These feature importance metrics can provide insights into how the model is making its predictions.

Ling's machine learning-assisted design of experiments proceeded by suggesting experiments with a high uncertainty in their result as modeled by the random forest regression. The intuition is that predictions by the response function which have low uncertainty have enough data to characterize the process-property relationships in that region of the design space. Therefore, researchers can have high confidence in the material properties they will achieve if parts are printed at those conditions. Thus, the process-property relationship likely has enough information to pick an optimal condition or investigate further. Regions of the design space with high uncertainty do not have enough information for the regression model. Thus, machine learning algorithms suggest these regions of the design space for further experiments. When only varying a few parameters at a time, regions of the design space which need characterization can be easily identified. When varying tens of parameters in additive manufacturing these regions of the design space are not apparent. Furthermore, correlated inputs can be masked by the complexity of the process-property relationship. Statistical models can spot these correlated regions of the design space.

A regression model, such as that used by Ling, is only one way of assessing process-property relationships. Random forests have been applied successfully to a range of applications in materials science. They have been used to discover new Heusler compounds [47] and new thermoelectric materials [67]. They have also been used to model material properties such as thermal conductivity in half-Heusler semiconductors [68] and to break down fields for dielectrics [69]. A review article detailing many optimization algorithms for design of experiments can be found in Shan et al. [70]. Adoption of machine-learning assisted design of experiments algorithms can rapidly increase the rate at which the relationship between AM process parameters and material properties are understood.

3. Topology Optimization

Alloy design and experimental design focus around combinatorial screening of inputs to either search for *new* properties or optimize on current properties. These optimizations reduce manufacturing cost, monetary or otherwise, and maximize performance capability. The same optimization can be applied to mechanical properties of parts. For structural materials, the goal is to optimize load bearing capacity or lifetime while minimizing the amount of material used. For aerospace, the goal is to minimize weight. Unique manufacturing geometries was one of the first intended applications of AM. Topology optimization (TO) focuses around exactly this task – finding optimized topological structures for a given mechanical application.

A filter is a mathematical operation which reveals information about a region of pixels/voxels in a mesh. Filters are most often represented as a product of a filter matrix with a matrix of mesh pixel values. Topology optimization proceeds by generating a CAD model of an AM part and modeling its performance, such as testing performance under mechanical load through an FEA simulation. Filters are applied to the CAD mesh which selectively removes material from the part. Then, the mechanical performance of the new part is modeled, followed by further material removal. This process proceeds until either a minimum weight/volume condition is met or the mechanical performance of the part is degraded.

In additive, topology optimization serves an additional purpose: TO algorithms can find un-printable regions of a part. Unsupported structures, low angle slopes, and certain part orientations during building are prohibited in AM because they will cause part deformation. An unsupported slope at acute angles can lead to part deformation and warpage [71]. Sacrificial support structures also need to be considered during topological optimization, along with the number of free-hanging features and the orientation of the part during manufacture. Langelaar et al. developed an AM-specific TO algorithm which searches for regions of parts that have too little support for manufacture [72, 73]. Other additive specific algorithms have been designed for optimizing density of parts [74]. These algorithms augment the AM process, both by taking advantage of the ability to optimize unique geometries, and also by identifying regions of parts which are incompatible with AM.

B. Process Design

Integrated computational materials engineering, as the name may imply, is primarily focused around *computational* engineering of materials. In experimental and engineering studies, the design space in AM makes choosing useful experiments difficult. Similarly, in modeling the wide range of AM physics to consider makes full-scale, full-physics computational difficult. This section focuses around applying machine learning algorithms to aid in computational studies and design of additive manufacturing.

1. Model Complexity and Dimensionality Reduction

Add high throughput grain boundary study of Homer [75]. A bottleneck in ICME approaches is the number of different models required to simulate all physics during additive manufacturing. The spatiotemporal scales that *could* be taken into consideration for AM process modeling span microns to meters. ICME of AM process modeling incorporates phase field, cellular automata, finite element, direct element methods, and more. At the smallest scale the feedstock, heat source, and melt pool dynamics have been modeled by finite element methods [2, 76, 77] or finite volume methods [78]. Microstructure growth from the melt pool is often the next phenomenon to be studied and has been modeled through phase field [4, 79– 81, cellular automata [3], or finite element methods [51]. Thermal histories of entire parts or sections can be modeled next, typically by finite element methods. Thermal history models look at heat transfer through the part [82], residual stress build up during manufacturing [5, 83], and thermal history such as cooling rate and temperature gradient [55, 84]. Full ICME approaches involve modeling the AM process at all these steps to determine the entire processing history for a part. Martukanitz et al. published a full ICME investigation of AM [85]. A review of ICME approaches across spatiotemporal scales can be found at [86] and a review of the physics of AM modeling can be found at [87]. A review of finite element methods specifically for AM can be found at [88].

For many engineering studies of AM not all of these physics need to be modeled or included. The complexity of AM, however, obscures which physics are relevant to accurately model a given final property. When considering the design space of additive, listing out the possible parameters and properties to model illustrates the complexity of modeling. A data

point in the AM process could be written as a vector

```
x = (Laser Power, Laser Speed, Laser Spot Size, ...
Energy Density, Alloy, Build Time, ...
Build Orientation, Hardness, Ultimate Tensile Strength, ...
Toughness, Porosity, Surface Roughness, ...).
```

By some estimates, the vector \mathbf{x} could be 160 variables long if only accounting for tunable machine parameters. Complexity in ICME of AM only grows when considering cross-correlated information across different models. The result of one simulation y could be an input \mathbf{x} of another simulation. A thermomechanical model may predict temperature gradient in deposited layers, while a phase field model may use thermal gradient to calculate solidification velocity. This problem motivates using computational models which balance accuracy with complexity.

Dimensionality reduction algorithms identify which parameters are relevant to model in an ICME approach and which are not, informing future ICME investigations of faster simulation routes to achieve the same result. Materials science has long had a need for dimensionally reduced, computationally accurate models. Some of the first applications of machine learning in materials science was for dimensionality reduction [24, 89–91]. The dimensionality reduction techniques covered in this review can be broadly classified into three mathematical frameworks: statistically driven methods, similarity analysis, and matrix factorization.

Statistically driven approaches can be employed to determine the parameters in **x** that strongly impact AM model outputs. A commonly used, statistically driven dimensionality reduction method is through random forest networks, as is done in design of experiments. The difference, however, is how uncertainty in the random forest is employed. In design of experiments, **outputs** with high associated uncertainty in the random forest model may be chosen for future tests. In dimensionality reduction, **inputs** with high uncertainty are indicative of irrelevant features.

Kamath utilized random forests to screen out irrelevant modeling parameters for predicting maximum density of additively manufactured parts [92]. Kamath started with an experimental dataset of manufacturing parameters and multiple modeling methods. An Eagar-Tsai simulation of a Gaussian laser beam on a powder bed was used to model ther-

mal conduction during manufacture. The model originally began with four inputs (laser power, speed, beam size, and powder absorptivity) and a design space of 462 possible input combinations. Kamath utilized random forests to determine which input was most important for achieving fully dense parts. If simulations are expensive to run then 462 different simulations may be out of the question. Identifying which parameters do not impact the final result reduces the size of input combinations, therefore reducing the number of computations to be performed. As a fictitious example, if the angle between the laser and part surface does not impact the result of a simulation, then simulations do not need to vary the angle. This is valuable, time-saving information in an ICME study.

Kamath identified that laser speed and power were the most important inputs out of the four to determine melt pool depth and shape. Thus, models that vary only these two parameters can be relied upon for their results. After determining the most important inputs, the same regression tree was applied in order to find optimized manufacturing conditions for fully dense parts, using the same approach. Instead of identifying which features impacted the model standard deviation, the machine settings which maximized y were found.

Another dimensionality reduction technique is focused around visualizing trends in datasets. Dimensionality reduction through similarity analysis allows visual interpretations of distributions across high dimensional spaces. Visualizing an n-dimensional distribution is difficult. Process maps are commonly employed in AM to visualize 2D slices of the AM design space [6]. Process maps are plots which chart the possible values of machine inputs and identify regions of the design space with similar properties. A commonly employed process map in AM of Ti-6Al-4V describes grain morphology as a function of solidification velocity R and temperature gradient G [93]. Extending process maps to n many process variables would require $\binom{n}{2}$ plots. Similarity analysis can be used to express trends in an n dimensional space in a human interpretable way without relying on multiple 2D process maps.

The 'similarity' in similarity analysis arises from the fact that models are trained on correlations between coordinates in the design space, instead of the coordinates themselves. These approaches combine similarity analysis with regression. Covariance between AM parameters are measured by similarity and these similarity scores are the data for a regression model.

t-distributed Stochastic Neighborhood Embedding (tSNE) is a dimensionality reduction

technique which measures distances in a high dimensional space and then projects data points onto a two dimensional plot. The similarity of all data points in the design space with each other is used to fit a distribution of similarities. The tSNE algorithm begins by fitting a probability distribution to all \mathbf{x} 's contained in a dataset. Relationships in n dimensional space are assessed through a kernel function $\kappa(\mathbf{x}, \mathbf{x}')$ which measures similarity between points in the design space. A commonly employed kernel is the Gaussian kernel

$$\kappa(\mathbf{x}, \mathbf{x}') = \frac{1}{\sqrt{2\pi\sigma^2}} \exp\left[-\frac{\mathbf{x} - \mathbf{x}'}{2\sigma^2}\right]$$
 (15)

where σ is a user-specified or fit standard deviation in the distribution of points in the design space. This kernel function assesses distance in the n dimensional space and assigns a similarity value between [0,1] depending on how similar the values are.

After the n dimensional dataset is fit, then a 2 dimensional coordinate \mathbf{x}^* is assigned to each \mathbf{x} . The reason for choosing a 2 dimensional coordinate is so that the final result can be visualized on a 2D plot. The tSNE algorithm fits a probability distribution to the n dimensional data set first, then assigns values to each \mathbf{x}^* such that they have the same probability as the associated high-dimensional \mathbf{x} . Once the probability distributions have been assigned, the \mathbf{x}^* values can be visualized on a 2D plot to investigate for trends.

The benefit of tSNE is that points which are close together in the n dimensional space appear close together on the 2 dimensional plot. This gives AM modelers an idea of how machine inputs and material behavior are distributed in the n dimensional space through a 2 dimensional visualization. This dimensionality reduction technique guides modelers as to what simulation conditions will reveal similar results. Furthermore, these 2D projections provide clusters of inputs and outputs which are similar in the high dimensional space. Analysis of these clusters can inform modelers as to correlations within their inputs and outputs, identifying which parameters can be screened out to save complexity.

A final dimensionality reduction technique requires expressing model data in a matrix and performing matrix factorization. For a dataset of model inputs and results, a matrix can be formed \mathbf{X} whose rows are coordinates in the design space \mathbf{x}_i . Matrix factorization techniques re-represent correlations in large datasets in a simplified way. The matrix $\mathbf{X}^T\mathbf{X}$ is a measure of covariance within \mathbf{X} , different than Eqn. 15 but equally valid. The matrix $\mathbf{X}^T\mathbf{X}$ can be very large due to the design space of additive manufacturing. One type of matrix factorization, called Principal Component Analysis (PCA) re-represents the data

matrix \mathbf{X} as

$$\mathbf{X} = \mathbf{U}\mathbf{\Sigma}^T \tag{16}$$

where the rows of Σ are the eigenvectors of $\mathbf{X}^T\mathbf{X}$ and the diagonal entries of \mathbf{U} are the square root of the eigenvalues $\sqrt{\lambda}$. PCA operates such that the first eigenvector in Σ indicates the most heavily correlated inputs of \mathbf{X} . The length of vectors in Σ is p many variables long. In cases where the original design space size n is large, the size of p is often much less than n, reducing the dimension of the problem. Regression can be performed on one, a few, or many of the eigenvectors in Σ to predict new model results using considerably less information than that contained in \mathbf{X} . Some studies have gone as far as to understand which correlations are being represented by the eigenvectors in Σ , revealing inputs or phenomenon which can be ignored during modeling. Such analysis is possible in AM, though requires further processing steps past matrix factorization. Materials science studies have utilized PCA previously to re-represent large datasets in simpler forms, such as predicting the formation energies of crystal structures from a lower dimensional space [94]. A review of applications of PCA in materials science can be found at [95].

In additive manufacturing, PCA can serve as a dimensionally reducing pre-processing technique. For a large data set with many machine parameters the design space can be reduced from vectors of n length to p length. The new vectors can then be used as regression model inputs for prediction of material properties based on trends observed in dataset \mathbf{X} .

2. Surrogate Modeling

Dimensionally reduced models are useful for studies that target specific phenomenon or properties. They are particularly useful for optimization. Modeling correlations between physical processes and the dynamics of AM is necessary to build better process-structure and process-property models. Phenomenon which are difficult to study experimentally, such as flow within the melt pool, are best studied through modeling approaches. Though expensive, full-physics modeling is quite necessary to understand how physics at different scales interact to impact the AM process. Full physics models, however, can be expensive and time consuming to run. If a model's computational expense is too high then performing simulations at all relevant manufacturing conditions can be impossible. Machine learning algorithms can use the results of previously run high fidelity simulations to fill in the gaps

and reduce development time.

A surrogate model is a regression model that interpolates the results of high-cost simulations. Surrogate models are regressed on the inputs and results of previously run simulations. Then, the surrogate model interpolates simulation results at new coordinates the design space. Surrogate models preclude the need for running computationally expensive simulations for every possible manufacturing condition. Surrogates can be as simple as linear regression between simulation inputs and results, but are often more complex. The accuracy of a surrogate model is dependent upon how many previous simulations have been run and at how many different points in the design space.

Tapia et al. built a surrogate model for laser powder bed fusion of 316L stainless steel. They were concerned with predicting the melt pool depth of single-track prints solely from the laser power, velocity, and spot size [96]. The dataset used to build the surrogate was computationally derived, based on previous simulation methods used by the same research team [97]. In particular, they used the results from a computationally expensive but high-accuracy melt pool flow model of Khairallah et al. [2]. They ran powder bed simulations at various laser powers, velocities, and spot sizes, and the model told them the depth of the melt pool, among other information. The datasets provided enough information for a surrogate model to be trained to predict simulation results.

To build surrogate model, Tapia et al. used a machine learning model known as a Gaussian process model (GPM). A common model assumption in Gaussian process modeling is

$$z(\mathbf{x}) = y(\mathbf{x}) + \epsilon(\mathbf{x}) \tag{17}$$

where $y(\mathbf{x})$ is the approximation (surrogate) of the simulation process, $\epsilon(\mathbf{x})$ is a stochastic, randomly distributed noise in measurement, and $z(\mathbf{x})$ is the value given by a simulation. GPMs make the assumption that all finite joint distributions of $y(\mathbf{x})$ are distributed in a multivariate normal manner, also known as a Gaussian distribution. For AM modeling this assumption holds if the model data used for a surrogate is stochastic in nature. The primary goal in GPMs is to find model parameters for the mean process $y(\mathbf{x})$ and a covariance function $C(\mathbf{x}, \mathbf{x}')$, which is a function of similar form to Eqn. ??. Fitting a Gaussian process model often begins with assuming a model function for covariance, fitting the model parameters to the observed values $z(\mathbf{x})$, then using those model parameters to predict simulation results $y(\mathbf{x})$ at other locations in the design space.

Starting with the covariance of their inputs, Tapia used Bayesian statistics to develop a probabilistic model that predicted melt pool depth from simulation inputs. They were able to successfully predict the outcomes of both high-fidelity simulations and experimental measurements solely by analyzing trends in previously obtained results. In particular, they were able to accurately predict the melt pool depth at a value that had never been observed before, either computationally or experimentally. For future investigations, predictions by the surrogate model can be relied up on instead of running a simulation or experiment.

Gaussian process models have benefits beyond their surrogate modeling capabilities. GPM provide robust uncertainty metrics on the predictions they make. Uncertainty in prediction is important in materials informatics because it aids in discriminating against poor prediction accuracy in machine learning. Some machine learning models do not have straightforward ways of assessing model error [98].

Another benefit of GPM is that it aids in inverse design and design space visualization. GPMs can explicitly identify regions of the design space which will maximize or minimize a value. In the case of Tapia et al. response surfaces were created from the GPM which visualized the depth of melt pools as a function of laser power and speed. Doing so allows engineers to identify regions of the design space which provide specific material responses, an important tool in optimization for additive.

Another approach to full scale models in AM is building high-fidelity models from an ensemble of low-fidelity models. Current integrated computational models link phenomena across spatiotemporal scales by running many single-physics models and passing the results from model to model, then comparing with experimental results. An example is the model of Martukanitz et al. that considered the thermal, mechanical, and material response of Ti-6Al-4V alloys manufactured with powder bed processes [85]. Martukanitz's model uses the finite element method to model the thermal and mechanical response based on classical continuum heat transfer equations. At the same time, the Calculation of Phase Diagrams (CALPHAD) method modeled the thermodynamics and mass transfer for each chemical species, while Diffusion-Controlled phase Transformations (DICTRA) is used to model phase formation. As noted by Martukanitz, this approach becomes computationally infeasible as the number of deposition layers increases.

Even a *single* modeling method – just FEA, phase field, CALPHAD, etc – becomes computationally expensive for thousands of layers of material deposition. Based on this

scaling, accurately modeling the physics of a full build seems prohibitive, at least without a major leap in computing power. A machine learning method known as *committee voting* or *ensemble modeling* may provide a workaround. These methods rely on sampling many individual models to predict the behavior of a larger class of physics.

In traditional ensemble modeling approaches, many different regression models are trained to model a relationship $y = f(\mathbf{x})$, as in regression trees. For a given input $\mathbf{x_1}$ all regression models are assessed and various outputs predicted, $f_1(\mathbf{x}), f_2(\mathbf{x}), ..., f_n(\mathbf{x})$, etc. Each model $f_i(\mathbf{x})$ is a different type of machine learning regression model. Different types of regression approaches are robust for different trends in data sets. The individual regression models could be a random forest, support vector machine, neural network, or any of the type of regression models discussed herein. A linear combination of these models can have a higher prediction accuracy than any individual model.

In AM, the same concept can be extended to datasets sampled from different physical models, instead of an ensemble of different regression models. Each model $f_i(\mathbf{x}_j)$ can be trained on the inputs and outputs of the jth simulation. These data sets for additive may be the inputs and solutions of a solidification or heat transfer model, for example, or experimentally obtained relationships.

Ensemble modeling can be used in AM to solve multiobjective optimization problems, such as optimizing the heat transfer through a build and the grain growth simultaneously. Doing so would require training an ensemble model of simulations. The simulations to include could be thermomechanical models of heat transfer, phase field models of grain growth, finite element models of laser absorption, and so forth. The final ensemble model would take the form

$$\mathbf{y}(\mathbf{x}) = \sum_{i=1}^{N} w_i f_i(\mathbf{x}) \tag{18}$$

where \mathbf{y} is a response vector of properties to optimize, $f_i(\mathbf{x})$ is a single simulation of the process out of N, and w_i are weights associated with each simulation. The vector \mathbf{y} would contain many parameters such as temperature T(t), temperature gradient ∇T , solidification velocity $\nu(t)$, and many more phenomenon to monitor from simulations.

In the AM case, where information from models may overlap, this type of approach can screen out noise from simulations or experiments to gauge a more fundamental relationship. The uncertainties associated with a model can be weighted by the experimentally observed

data points. Furthermore, predictions can be made across many different physical models, resulting in a predictive method for holistic analysis of the final properties. This method does not provide a picture of how the physics in each simulation interact or disagree. However, it can be used to simultaneously optimize the results of many different simulations and point toward desirable manufacturing conditions. A review of multiobjective optimization functions can be found at [99].

Meredig et al. applied this method to the prediction of ABC ternary compounds, where A, B, and C each represent an element [100]. The space of all combinations of A, B, and C elements is large and would require many, many simulations. Instead of running high-throughput DFT for all possibilities they ran calculations for AB, AC, and BC compounds to generate a database of information. Then, regression trees were trained from the inputs of each binary alloy simulation to predict the formation energy. Finally, an ensemble model was trained

$$E(ABC) = w_{AB}f(\mathbf{x}_{AB}) + w_{AC}f(\mathbf{x}_{AC}) + w_{BC}f(\mathbf{x}_{BC})$$
(19)

where E() is the formation energy, w_{AB} is the weight for a regression model of AB alloys, $f(\mathbf{x}_{AB})$ is the regression tree for all AB alloys, and so on. In the end, Meredig's model was able to predict 4,500 new, stable ternary materials.

Machine learning is not only limited to ex situ experimental investigations or modeling approaches. Machine learning has also made major advances in signal processing and feedback. Many of the same ideas which apply to experimental and modeling studies can also be aids for in situ analysis and feedback of the manufacturing process.

C. Process Monitoring and Characterization

Computer vision is a class of image recognition algorithms that have been developed for automated feature identification in signals. Intelligent computer vision utilizes machine learning algorithms to identify objects and features in images and time-series data.

Monitoring of AM processes produces data equally numerous to, if not in excess of, the data produced by parametric analysis and modeling. The numerousness of time series data in AM warrants usage of quick, efficient, and robust signal processing methods for process monitoring, feedback, and control. These signal processing algorithms are closely related to machine learning. They serve as tools in their own right, and can also pre-process data for

use in other machine learning applications, like clustering and regression.

1. In Situ Process Monitoring and Feedback

Need to include Kwon2018 [101] – paper about using a NN for to classify melt-pool images.

Thus far, in situ control in AM has been consistently ranked as one of the most-needed technologies for advancing the technology [102–104]. The combination of rapid solidification and the small length scales of AM solidification can make traditional process monitoring approaches difficult. Machine learning can fill in gaps where human-specified process monitoring models are insufficient.

Computer vision can be employed in additive manufacturing to monitor the printing process, such as characterizing temperature profiles, identifying abnormal melt pool morphologies, and automatically detecting defect formation. Doing so will require methods for in situ process monitoring and data collection.

Process monitoring involves acquisition of realtime signals which can reveal information about a wide variety of phenomenon during manufacturing. McKeown et al. used dynamic transmission electron microscopy to measure solidification rates in powder bed AM [105]. Bertoli et al. also characterized cooling rates using high speed imaging [106]. Raplee et al. used thermography to monitor the solidification and cooling rates of electron beam powder bed fusion, then related the temperature profiles to defect and microstructural characteristics [107]. Distortion of parts due to thermal cycling was investigated by Denlinger et al. by means of thermocouples in contact with the build substrate [61]. All of these methods are amenable to aid by computer vision and signal processing.

The type of data being collected in situ is most often in the form of time series data or images. Signal processing of either typically involves identifying signals which deviate from a mean, desired signal. Examples include a spike in temperature from a thermocouple or a sharp change in intensity in an image, both of which may indicate a deviation from the desired processing conditions. Image processing tools known as *filters* can be used to identify signals of interest in AM data. From a mathematical perspective, filters are implemented for signal/image processing in the same way as for topology optimization.

A filter is implemented as mathematical operation on a segment of time series data or an

area of pixels in an image. The simplest filters identify characteristics of the pixel intensity in that region and highlight them. An example for an image would be to identify regions of sharp change in intensity, which often correspond to boundaries between objects. An example would be the change in intensity between a liquid melt pool and the powder bed. Signals of interest or regions of interest in an image identified by filters are known as features.

The use of filters alone does not constitute machine learning, but filters can be used in regression algorithms to learn signal features which are indicative of behavior in the manufacturing process. Defect detection is an apt use of filters in a machine learning approach. Repeated defect-free printing involves following the same processing parameters every manufacture. Regression algorithms can predict when a defect appears by monitoring when features indicative of defects are found by filters.

A computer vision method called template matching can be utilized for automatic identification of signal features. Template matching involves comparison of a database of preidentified signal features with the features measured in situ. The scale-invariant feature
transform (SIFT) [108] and a variant of it, speeded-up robust features (SURF) [109] are
both feature identification algorithms which can be used for template matching. A specific
type of template matching is the 'bag of visual words' or 'dictionary' method. A collection
(dictionary) of typical features from the AM process can be built based on features obtained
from filters. The features measured in situ are compared with dictionary entries. If an in
situ feature matches a defect-indicative feature from the dictionary, then it is likely a defect
has formed in manufacturing process.

Template matching is carried out through similarity of analysis of in situ features and dictionary features. Outputs of filters can also be used as inputs to regression algorithms to predict the probability of a feature being present in an image. For example, an algorithm may be trained which convolves three different filters with a signal acquired in situ. Each filter likely highlights a different aspect of the signal. The goal of the algorithm is to predict whether or not a defect is present in pixel location P(x, y). A regression algorithm can be trained

$$P(x,y) = w_1 I_1(x,y) + w_2 I_2(x,y) + w_3 I_3(x,y)$$
(20)

where the probability of a defect being present is dependent upon the results of filter operations on regions of the image. This type of algorithm is the basis for a *neural network* where filters are used as regression functions for feature identification in images. Recently,

advances in computer vision has led to the advent of *convolutional neural networks* which have high prediction accuracy for feature identification in images.

Convolutional neural networks not only adjust the weights w_i of regression algorithms for feature identification, but also adjust the filters themselves. That is, convolutional neural networks 'learn' the best filters for elucidating features which are indicative of process phenomenon. CNNs use large databases of labeled information to learn features in the image that indicate whether or not certain human-identified objects are present. Convolutional neural networks can identify multiple features or defects in an image simultaneously. A downside to CNNs is that they require **very** large datasets (thousands of images, at the least) of labeled data to be successful. Neural networks have already been implemented for in situ AM analysis. Yuan et al [110] were able to successfully monitor melt track width, standard deviation, and continuity of tracks in situ during laser powder bed manufacturing. Scime and Beuth trained a convolutional neural network to identify six different types of defect which are typical of laser powder bed fusion, with fairly good prediction accuracy [111]. These studies encompass only a few of the possibilities of in situ process monitoring for AM.

2. Featurization of Qualitative Image Data

New article by Miyazaki et al. on segmentation of SLM Ti-6Al-4V; they are able to segment out the alpha and beta phases, characterize fractions, etc. [112]. Also you can cite Bulgarevich [113] but you don't need to go in-depth on his paper, lump it in with the DeCost stuff. The use of images in studying additive manufacturing is widespread, common throughout all aspects of the manufacturing process, and provides key information about material properties and processing. As with all aspects of AM, the sheer size of image data to be analyzed is profound due to the large design space of AM. The types of information taken from images includes grain characteristics, like size, orientation, and phase, and defect characteristics, like pore size or crack length. When characterizing all of these features for all possible processing conditions and alloys the size of the problem grows quickly.

Computer vision algorithms have been tested for automation of materials science image classification and analysis. Using these algorithms can speed up the experimental characterization process of AM. Furthermore, computer vision techniques can quantify information which may have otherwise only been used qualitatively or measured by approximation.

It is worthwhile to mention up front that these algorithms have been *tested* on microstructure and, in some cases, additive-specific images. There are few algorithms that can process AM microstructure data 'out-of-the-box.' Rather, these algorithms will need to be tailored in order to quantify AM images specifically. However, the algorithms discussed here have been proven on non-AM microstructure datasets, thus they should be extensible to AM datasets. The computer vision approaches which work for microstructure data are often the same approaches discussed in the previous section for in situ monitoring.

One AM-related application of image characterization is measuring particle size distributions in AM powder feedstock. DeCost and Holm used SIFT with a dictionary classifier, as in template matching, to measure the particle size distribution for a dataset of synthetic powder particles [114]. Particle size distribution plays in several steps across the additive process including energy absorption and part metrology [13, 115, 116]. DeCost created datasets with six different particle size distributions. Image features were identified and classified using k-means clustering on the features found by SIFT. Then, a classification algorithm known as a support vector machine (SVM) was trained to classify image features into particle sizes. DeCost was able to achieve 89% overall classification accuracy in measuring particle size distribution this way. DeCost et al. later improved upon this powder classification method and were able to achieve higher classification accuracies for real powder images [117].

DeCost et al. have also made strides in identifying and quantifying information from metallographs [63, 118, 119]. A good portion of quality control in materials science as a whole, not just AM, involves classifying materials based on metallographs or micrographs of microstructure. Work is being done across materials science to apply machine learning based computer vision to classifying and quantifying information in these microstructural images. Doing so will speed up the process of materials characterization and qualification, while also providing methods of quantifying information which otherwise would have stayed in a qualitative form. Examples include classification of grain stuctures, measurements of grain size, pore size calculations, and more.

Chowdhury et al. took a more expansive approach to performing feature identification in microstructures. In particular, they were looking to classify microstructures as either dendritic or non dendritic. Chowdhury employed 8 different feature identification methods for a dataset of images. Classification was performed using an ensemble of ML techniques including support vector machines (SVM), Naïve Bayes, nearest neighbor, and a committee of the three previous classification methods [120]. Chowdhury's wide approach to image classification compared the predictive ability of all combinations of feature identification and classification methods, achieving classification accuracies above 90%.

It would be overly burdensome to lay out every *possible* application of computer vision in additive manufacturing. Efforts are underway across materials science to implement computer vision for the automation of materials classification. Rather, the authors would like to refer the reader to reviews on the subject of computer vision for materials science, as well as open libraries. The hope is that readers will discover the many possible uses of computer vision and begin applying methods to their own AM problems.

IV. LEARNING FROM THE PAST: MOVING TOWARDS DATABASE-DRIVEN DESIGN OF ADDITIVE

The scientific approaches to studying additive manufacturing discussed herein – parametric analysis, computational modeling, in situ monitoring, and the like – produce data. The application of machine learning to these scientific approaches likewise produces data. All of this data contains a slice of the AM design space. AM researchers who have access to more data in the design space can better tailor manufacturing processes and material properties. Furthermore, machine learning algorithms trained on large datasets typically have higher prediction accuracy and a greater range of behaviors they can model. Making AM process-structure and process-property data open and accessible to the scientific public accelerates the rate at which research and engineering advances.

Databases of process-structure and process-property relationships are not a new concept in materials science. Databases like the Linus Pauling Files or International Crystal Structure Database have been widely used for materials design. However, databases of materials information are beginning to expand beyond general materials properties like thermodynamic behavior or crystal structure. Domain-specific databases are generated from high throughput experimental and computational investigations that have occurred over the past thirty years. Studying and understanding the development of data-driven fields in materials science can aid in the development of data-driven approaches for additive manufacturing.

Experimental high throughput investigations have been used in materials science for many decades [121]. Common deposition techniques have enough degrees of freedom to allow for continuous compositional variation within a single sample, which allows for continuous mapping of composition-structure-property relationships [122–124]. These manufacturing methods have the same design space problem that additive has: the number of possible input combinations obscures many of the important underlying process-structure phenomenon. In cases like this, manufacturing and characterizing large catalogues of samples can lead to optimized properties faster than a theory-driven approach [125, 126]. High throughput deposition studies with chemical vapor deposition, metallorganic chemical vapor deposition, physical vapor deposition, and atomic layer deposition, among other techniques are commonplace for the manufacturing of sensors, batteries, photovoltaics, electronics, and the like [33, 127–129]. Furthermore, the parameters of interest in these studies can sometimes be quickly catalogued using high throughput characterization techniques like laboratory X-ray diffraction and electron probe microanalysis.

These studies culminate in large libraries of material properties listed as a function of composition. As far back as the 1990s, data-driven algorithms were being applied to these large libraries of composition-property data. Evolutionary and genetic algorithms were trained on composition to predict stable crystal structure and material properties [31, 35, 38, 130, 131]. Even neural networks, which did not have the widespread then use that they have now, were being applied for the prediction of crystal structures based on composition [132].

Modeling challenges in materials science have also been tackled using large databases with machine learning. Packages such as the Vienna Ab initio Simulation Package (VASP) have been employed for high throughput searches of stable material systems with a wide range of properties. Computational high throughput investigations have proven to closely match experiments in many regimes [133]. High throughput density functional theory (DFT) studies generate quite a bit of data and are therefore well equipped for machine learning and database-driven design. The application of high throughput DFT is widespread for design of materials with all sorts of properties including high temperature superconductors [134], lithium ion batteries [29, 135, 136], cathode materials [137], piezoelectrics [32], ferroelectrics [138], corrosion resistant films [139], and thermoelectrics [140, 141]. Each of these studies, like parametric studies in additive, vary a set number of model input parameters and measure

a material property as the dependent response.

Yet many of the same modeling obstacles exist in DFT as in AM, such as a lack of transferability between models and the computational expense of large material systems. The design space problem exists here as well – there are so many possible compositional combinations that knowing where to look is difficult. Machine learning was proposed as a solution for obstacles in high throughput DFT as early as 2005 [142]. Large unit cells whose properties cannot be directly calculated using DFT are often approximated using machine learning approaches like neural networks [143], genetic algorithms [39], and principal component analysis [91]. Studies applying machine learning to databases of computational information have gone beyond tackling computational problems. In some cases, the studies have revealed previously unobserved or uncharacterized relationships between crystal structure information and materials properties [144].

Image processing databases have developed outside of materials science specifically, though a database of materials-specific images is highly desirable for the training of recognition algorithms. The collection and distribution of image databases have enabled rapid developments in the field of computer vision. Many of the more common objectives with computer vision – autonomous navigation, face recognition, object recognition, image segementation – have databases which are catalogued in online repositories like CVonline [145] and VisionScience [146]. Open sharing of additive microstructures will aid in the development of segmentation and identification algorithms which are suited for additive specific problems.

In an effort to reduce the design time on material systems, programs like the Materials Project incorporate data taken from a wide range of experimental and computational methods into an open-source, accessible database. The Materials Project also features electronic, structural, and thermodynamic calculations of different materials as well as an automated workflow for doing DFT computations of material systems [22, 147]. Other databases of materials information include AFLOWLib [21, 148], the Harvard Clean Energy Project [149], Japan's National Institute of Material Science [150], and the Open Quantum Materials Database [151]. Some pipelines for high-throughput computation and analysis have included consideration of publication timelines in their processes [152]. These databases offer a multitude of benefits to materials researchers. First and foremost, publicly accessible databases offer an infrastructure for the free flow of experimental and computational results.

Synergy between research groups becomes easier as data is shared more freely.

Furthermore, many of these online databases also provide tools for performing material design. The Materials Project offers a design interface, whereby users can specify a set of material properties and are provided with a list of likely candidate materials. Other projects, like AFLOW, allow for fast high-throughput DFT calculations of a wide range of material systems.

Having open, accessible databases also improves the rate at which machine learning can be applied to design for additive manufacturing. Machine learning as a tool driving materials design was proposed some time ago. Review articles have explored the many and varied uses of machine learning across materials science, many of the applications finding great success [153–155]. A review article on best practices for machine learning in materials science can be found here [156].

The generation of databases that are accessible to the scientific public is a primary step on the roadmap of the Materials Genome Initiative [157]. Much of the development of materials databases have focused on computationally-derived materials information. Infrastructure and standards need to be developed that allow for sharing of experimental data that is understandable and usable by many researchers. Data journals are becoming more common for sharing datasets from scientific investigations and are making strides in standardizing data-sharing infrastructure [158]. The publication of datasets themselves for public use is becoming more common [159, 160].

Additive manufacturing should move toward the same types of infrastructure for open data sharing. The combinatorial problems in additive are widespread and cover many, many length scales. Large institutions may have the resources to link time- and length-scales in additive manufacturing. Smaller research groups are often limited to studying a single process phenomenon and do not necessarily have means to integrate their knowledge into other additive manufacturing studies. The generation of additive databases allows for a democratization of research and an acceleration of the pace at which additive manufacturing advances are made.

V. CONCLUSIONS

- [1] "National science and technology council, office of science and technology policy, materials genome initiative for global competitiveness," (2011).
- [2] S. A. Khairallah, A. T. Anderson, A. Rubenchik, and W. E. King, Acta Materialia 108, 36 (2016).
- [3] W. Tan, N. S. Bailey, and Y. C. Shin, Computational Materials Science 50, 2573 (2011).
- [4] J. Kundin, L. Mushongera, and H. Emmerich, Acta Materialia 95, 343 (2015).
- [5] D. Pal, N. Patil, K. Zeng, and B. Stucker, Journal of Manufacturing Science and Engineering 136, 1 (2014).
- [6] J. Beuth and N. Klingbeil, Journal of Materials: Laser Processing (2001).
- [7] J. Quinlan, Machine Learning 1, 81 (1986).
- [8] J. B. Tenenbaum, V. de Silva, and J. C. Langford, Science 290 (2000).
- [9] S. T. Roweis and L. K. Saul, Science **290** (2000).
- [10] R. Bro and A. K. Smilde, Analytical Methods Tutorial Review 6 (2014).
- [11] J. J. Grefenstette, IEEE Transactions of Systems, Man, and Cybernetics 16 (1986).
- [12] Z.-K. Liu, L.-Q. Chen, and K. Rajan, Journal of Materials: Integrated Computational Materials, 42 (2006).
- [13] C. Boley, S. Mitchell, A. Rubenchik, and S. Qu, Applied Optics 55, 6496 (2016).
- [14] J. Trapp, A. M. Rubenchik, G. Guss, and M. J. Matthews, Applied Materials Today 9, 341 (2017).
- [15] G. Bi, C. N. Sun, and A. Gasser, Journal of Materials Processing Technology 213, 463 (2013).
- [16] J. H. Martin, B. D. Yahata, J. M. Hundley, J. A. Mayer, T. A. Schaedler, and T. M. Pollock, Nature Letters 549, 365 (2017).
- [17] P. Collins, D. Brice, P. Samimi, I. Ghamarian, and H. Fraser, Annual Review of Materials Research 46, 63 (2016).
- [18] C. A. Brice, W. A. Tayon, J. A. Newman, M. V. Kral, C. Bishop, and A. Sokolova, Materials Characterization (2018).

- [19] A. Wang, S. Song, Q. Huang, and F. Tsung, IEEE Transactions on Automation Science and Engineering 14, 968 (2017).
- [20] P. Villars, N. Onodera, and S. Iwata, Journal of Alloys and Compounds 279, 1 (1998).
- [21] S. Curtarolo, W. Setyawan, S. Wang, J. Xue, K. Yang, R. H. Taylor, L. J. Nelson, G. L. W. Hart, S. Sanvito, M. Buongiorno-Nardelli, N. Mingo, and O. Levy, Computational Materials Science 58, 227 (2012).
- [22] A. Jain, S. P. Ong, G. Hautier, W. Chen, W. D. Richards, S. Dacek, S. Cholia, D. Gunter,
 D. Skinner, G. Ceder, and K. A. Persson, APL Materials 1 (2013), 10.1063/1.4812323.
- [23] A. Fransceschetti and A. Zunger, Letters to Nature 402 (1999).
- [24] C. C. Fischer, K. J. Tibbetts, D. Morgan, and G. Ceder, Nature Materials 5, 641 (2006).
- [25] A. R. Oganov and C. W. Glass, Journal of Chemical Physics 124 (2006), 10.1063/1.2210932.
- [26] Y. Ikeda, Materials Transactions **38**, 771 (1997).
- [27] A. M. Gopakumar, P. V. Balachandran, D. Xue, J. E. Gubernatis, and T. Lookman, Nature Scientific Reports 8 (2018).
- [28] D. Wu, Y. Tian, L. Zhang, Z. Wang, J. Sheng, W. Wang, K. Zhou, and L. Liu, Materials 11 (2018).
- [29] S. Kirklin, B. Meredig, and C. Wolverton, Advanced Energy Materials 3, 252 (2013).
- [30] W. Setyawan, R. M. Gaume, S. Lam, R. S. Feigelson, and S. Curtarolo, ACS Combinatorial Science, 382 (2011).
- [31] D. Wolf, O. Buyevskaya, and M. Baerns, Applied Catalysis A: General 200, 63 (2000).
- [32] A. Roy, J. W. Bennett, K. M. Rabe, and D. Vanderbilt, Physical Review Letters 109, 1 (2012), arXiv:1107.5078.
- [33] G. Gilmer, H. Huang, and C. Roland, Computational Materials Science 12, 354 (1998).
- [34] H. Koinuma and I. Takeuchi, Nature Materials 3, 429 (2004).
- [35] J. Morris, D. Deaven, and K. Ho, Physical Review B 53, R1740 (1996).
- [36] K.-M. Ho, A. A. Shvartsburg, B. Pan, Z.-Y. Lu, C.-Z. Wang, J. G. Wacker, J. L. Fye, and M. F. Jarrold, Nature (London) 392, 582 (1998).
- [37] G. H. Jóhannesson, T. Bligaard, A. V. Ruban, H. L. Skriver, K. W. Jacobsen, and J. K. Norskov, Physical Review Letters 88, 2555061 (2002).
- [38] D. P. Stucke and V. H. Crespi, Nano Letters 3, 1183 (2003).
- [39] G. L. W. Hart, V. Blum, M. J. Walorski, and A. Zunger, Nature Materials 4, 391 (2005).

- [40] A. Kulkarni, K. Krishnamurthy, S. Deshmukh, and R. Mishra, Materials Science and Engineering A 372, 213 (2004).
- [41] W. Xu, P. R.-D. del Castillo, and S. van der Zwaag, Philosophical Magazine 88, 1825 (2008).
- [42] S. Dudiy and A. Zunger, Physical Review Letters 97 (2006).
- [43] S. M. Anijdan, A. Bahrami, H. M. Hosseini, and A. Shafyei, Materials and Design 27, 605 (2006).
- [44] G. Hautier, C. C. Fischer, A. Jain, T. Mueller, and G. Ceder, Chemistry of Materials 22, 3762 (2010).
- [45] N. Chakraborti, International Materials Reviews 49, 246 (2004).
- [46] G. A. Landrum and H. Genin, Journal of Solid State Chemistry 176, 587 (2003).
- [47] A. Oliynyk, E. Antono, T. Sparks, L. Ghadbeigi, M. Gaultois, B. Meredig, and A. Mar, Chemistry of Materials 28, 7324 (2016).
- [48] A. Antonysamy, J. Meyer, and P. Prangell, Materials Characterization 84, 153 (2013).
- [49] G. Strano, L. Hao, R. M. Everson, and K. E. Evans, Journal of Material Processing Technology 21, 589 (2013).
- [50] S. Bontha, N. W. Klingbeil, P. A. Kobryn, and H. L. Fraser, Materials Science and Engineering A 513-514, 311 (2009).
- [51] P. Nie, O. Ojo, and Z. Li, Acta Materialia 77, 85 (2014).
- [52] Y. Chen, F. Lu, K. Zhang, P. Nie, S. R. E. Hosseini, K. Feng, and Z. Li, Journal of Alloys and Compounds 670, 312 (2016).
- [53] B. Baufeld, E. Brandl, and O. Van Der Biest, Journal of Materials Processing Technology 211, 1146 (2011).
- [54] S. Bontha, N. W. Klingbeil, P. A. Kobryn, and H. L. Fraser, Journal of Materials Processing Technology 178, 135 (2006).
- [55] Y. Li and D. Gu, Materials and Design **63**, 856 (2014).
- [56] J. Cherry, H. M. Davies, S. Mehmood, N. P. Lavery, S. G. R. Brown, and J. Sienz, International Journal of Advanced Manufacturing Technology 76, 869 (2015).
- [57] Q. Jia and D. Gu, Journal of Alloys and Compounds **585**, 713 (2014).
- [58] J. Delgado, J. Ciurana, and C. A. Rodriguez, International Journal of Advanced Manufacturing Technology **60**, 601 (2012).
- [59] A. M. Khorasani, I. Gibson, U. S. Awan, and A. Ghaderi, Additive Manufacturing (2018).

- [60] A. S. Wu, D. W. Brown, M. Kumar, G. F. Gallegos, and W. E. King, Metallurgical and Materials Transactions A: Physical Metallurgy and Materials Science 45, 6260 (2014).
- [61] E. R. Denlinger, J. C. Heigel, P. Michaleris, and T. Palmer, Journal of Materials Processing Technology 215, 123 (2015).
- [62] C. E. Shannon, The Bell System Technical Journal 27, 379 (1948), arXiv:9411012 [chao-dyn].
- [63] J. Ling, M. Hutchinson, E. Antono, B. DeCost, E. A. Holm, and B. Meredig, Materials Discovery 10, 19 (2017).
- [64] B. Efron, Journal of the Royal Statistical Society Series B (Methodological), 83 (1992).
- [65] B. Efron, Journal of the American Statistical Association 109, 991 (2014).
- [66] S. Wager, T. Hastie, and B. Efron, Journal of Machine Learning Research 15, 1625 (2014), arXiv:1311.4555v2.
- [67] M. Gaultois, A. Oliynyk, A. Mar, T. Sparks, G. Mulholland, and B. Meredig, APL Materials 4, 053213 (2016).
- [68] J. Carrete, W. Li, N. Mingo, S. Wang, and S. Curtarolo, Phys. Rev. X 4, 011019 (2014).
- [69] C. Kim, G. Pilania, and R. Ramprasad, Chemistry of Materials 28, 1304 (2016).
- [70] S. Shan and G. G. Wang, Structural and Multidisciplinary Optimization 41, 219 (2010).
- [71] A. T. Gaynor and J. K. Guest, Structural and Multidisciplinary Optimization 54, 1157 (2016).
- [72] M. Langelaar, Additive Manufacturing 12, 60 (2016).
- [73] M. Langelaar, Structural and Multidisciplinary Optimization 55, 871 (2017).
- [74] T. Zegard and G. H. Paulino, Structural Multidisciplinary Optimization 53, 175 (2016).
- [75] E. R. Homer, Computational Materials Science 161, 244 (2019).
- [76] E. Toyserkani, A. Khajepour, and S. Corbin, Optics and Lasers in Engineering 41, 849 (2004).
- [77] V. Manvatkar, A. De, and T. DebRoy, Journal of Applied Physics 116 (2014).
- [78] D. Dai and D. Gu, Materials and Design 55, 482 (2014).
- [79] L.-Q. Chen, Annual Review of Materials Research 32, 113 (2002).
- [80] X. Gong and K. Chou, Journal of Materials 67, 1176 (2015).
- [81] S. Sahoo and K. Chou, Additive Manufacturing 9, 14 (2016).
- [82] P. Michaleris, Finite Elements in Analysis and Design 86, 51 (2014).
- [83] J. Ding, P. Colegrove, J. Mehnen, S. Ganguly, P. S. Almeida, F. Wang, and S. Williams,

- Computational Materials Science **50**, 3315 (2011).
- [84] N. Raghavan, R. Dehoff, S. Pannala, S. Simunovic, M. Kirka, J. Turner, N. Carlson, and S. S. Babu, Acta Materialia 112, 303 (2016).
- [85] R. Martukanitz, P. Michaleris, T. Palmer, T. DebRoy, Z.-K. Liu, R. Otis, T. W. Heo, and L.-Q. Chen, Addtive Manufacturing 1, 52 (2014).
- [86] M. Francois, A. Sun, W. King, N. Henson, D. Tourret, C. Bronkhorst, N. Carlson, C. Newman, T. Haut, J. Bakosi, J. Gibbs, V. Livescu, S. Vander Wiel, A. Clarke, M. Schraad, T. Blacker, H. Lim, T. Rodgers, S. Owen, F. Abdeljawad, J. Madison, A. Anderson, J.-L. Fattebert, R. Ferencz, N. Hodge, S. Khairallah, and O. Walton, Current Opinion in Solid State and Materials Science, 1 (2017).
- [87] W. King, A. Anderson, R. Ferencz, N. Hodge, C. Kamath, and S. Khairallah, Materials Science and Technology 31 (2015).
- [88] M. Gouge, P. Michaleris, E. Denlinger, and J. Irwin, Thermo-Mechanical Modeling of Additive Manfuacturing, Chapter 2: The Finite Element Method for the Thermo-Mechanical Modeling of Additive Manufacturing Processes (Elsevier Inc, 2018).
- [89] J. A. Flores-Livas, A. Sanna, and S. Goedecker, Novel Superconducting Materials 3, 6 (2017).
- [90] M. Rupp, A. Tkatchenko, K.-R. Müller, and O. A. von Lilienfeld, Physical Review Letters 208, 1 (2011), arXiv:1109.2618.
- [91] J. C. Snyder, M. Rupp, K. Hansen, K. R. Müller, and K. Burke, Physical Review Letters 108, 1 (2012), arXiv:1112.5441.
- [92] C. Kamath, International Journal of Advanced Manufacturing Technology 10 (2016).
- [93] R. R. Dehoff, M. M. Kirka, W. J. Sames, H. Bilheux, A. S. Tremsin, L. E. Lowe, and S. S. Babu, Materials Science and Technology 31 (2015).
- [94] S. Curtarolo, D. Morgan, K. Persson, J. Rodgers, and G. Ceder, Physical Review Letters 91, 135503 (2003), arXiv:0307262 [cond-mat].
- [95] K. Rajan, C. Suh, and P. F. Mendez, Statistical Analysis and Data Mining 1, 361 (2009).
- [96] G. Tapia, S. A. Khairallah, M. Matthews, and W. E. King, International Journal of Advanced Manufacturing Technology 10 (2017).
- [97] W. E. King, H. D. Barth, V. M. Castillo, G. F. Gallegos, J. W. Gibbs, D. E. Hahn, C. Kamath, and A. M. Rubenchik, Journals of Materials Processing Technology 214, 2915 (2014).
- [98] M. A. Bessa, R. Bostanabad, Z. Liu, A. Hu, D. W. Apley, C. Brinson, W. Chen, and W. K.

- Liu, Computer Methods in Applied Mechanics and Engineering 320, 633 (2017).
- [99] Y. Jin and B. Sendhoff, IEEE Transactions of Systems, Man, and Cybernetics Parc C: Applications and Reviews 38 (2008).
- [100] B. Meredig, A. Agrawal, S. Kirklin, J. E. Saal, J. W. Doak, A. Thompson, K. Zhang, A. Choudhary, and C. Wolverton, Physical Review B - Condensed Matter and Materials Physics 89, 1 (2014).
- [101] O. Kwon, H. G. Kim, M. J. Ham, W. Kim, G.-H. Kim, J.-H. Cho, N. I. Kim, and K. Kim, Journal of Intelligent Manufacturing (2018).
- [102] S. Berumen, F. Bechmann, S. Lindner, J.-P. Kruth, and T. Craeghs, Physics Procedia 5, 617 (2010).
- [103] G. Tapia and A. Elwany, Journal of Manufacturing Science and Engineering 136, 060801 (2014).
- [104] M. Mani, B. M. Lane, M. A. Donmez, S. C. Feng, and S. P. Moylan, International Journal of Production Research 55 (2017).
- [105] J. T. McKeown, K. Zweiacker, C. Liu, D. R. Coughlin, A. J. Clarke, J. K. Baldwin, J. W. Gibbs, J. D. Roehling, S. D. Imhoff, P. J. Gibbs, D. Tourret, J. M. Wiezorek, and G. H. Campbell, JOM 68 (2016).
- [106] U. S. Bertoli, G. Guss, S. Wu, M. J. Matthews, and J. M. Schoenung, Materials and Design 135, 385 (2017).
- [107] J. Raplee, A. Plotkowski, M. M. Kirka, R. Dinwiddie, A. Okello, R. R. Dehoff, and S. S. Babu, Scientific Reports 7, 1 (2017).
- [108] D. G. Lowe, International Journal of Computer Vision 60, 91 (2004).
- [109] H. Bay, A. Ess, T. Tuytelaars, and L. V. Gool, Computer Vision and Image Understanding 110, 346 (2008).
- [110] B. Yuan, G. M. Guss, A. C. Wilson, S. P. Hau-Riege, P. J. DePond, S. McMains, M. J. Matthews, and B. Giera, Adv. Mater. Technol., 1 (2018).
- [111] L. Scime and J. Beuth, Additive Manufacturing 24, 273 (2018).
- [112] S. Miyazaki, M. Kusano, D. S. Bulgarevich, S. Kishimoto, A. Yumoto, and M. Watanabe, Materials Transactions (2019).
- [113] D. S. Bulgarevich, S. Tsukamoto, T. Kasuya, M. Demura, and M. Watanabe, Scientific Reports 8, 1 (2018).

- [114] B. L. DeCost and E. A. Holm, Computational Materials Science 126, 438 (2017).
- [115] J. Zhou, Y. Zhang, and J. Chen, Journal of Manufacturing Science and Engineering 131, 1 (2009).
- [116] C. Boley, S. Khairallah, and A. Rubenchik, Applied Optics 54, 2477 (2015).
- [117] B. L. DeCost, H. Jain, A. D. Rollett, and E. A. Holm, Jom 69, 456 (2017).
- [118] B. L. Decost and E. A. Holm, Computational Materials Science 110, 126 (2015).
- [119] B. L. DeCost, T. Francis, and E. A. Holm, Acta Materialia 133, 30 (2017), arXiv:1702.01117.
- [120] A. Chowdhury, E. Kautz, B. Yener, and D. Lewis, Computational Materials Science 123, 176 (2016).
- [121] X. D. Xiang, X. Sun, G. Briceno, Y. Lou, K.-A. Wang, H. Chang, W. G. Wallace-Freedman, S.-W. Chen, and P. G. Schultz, Science 268, 1738 (1995).
- [122] C. J. Long, J. Hattrick-Simpers, M. Murakami, R. C. Srivastava, I. Takeuchi, V. L. Karen, and X. Li, Review of Scientific Instruments 78 (2007), 10.1063/1.2755487.
- [123] C. J. Long, D. Bunker, X. Li, V. L. Karen, and I. Takeuchi, Review of Scientific Instruments 80 (2009), 10.1063/1.3216809.
- [124] A. G. Kusne, T. Gao, A. Mehta, L. Ke, M. C. Nguyen, K.-M. Ho, V. Antropov, C.-Z. Wang, M. J. Kramer, C. Long, and I. Takeuchi, Scientific Reports 4, 6367 (2015).
- [125] G. Ceder, Y.-M. Chiang, D. R. Sadoway, M. K. Aydinol, Y.-I. Jang, and B. Huang, Nature 392, 694 (1998).
- [126] G. Pilania, C. Wang, X. Jiang, S. Rajasekaran, and R. Ramprasad, Scientific Reports 3, 2810 (2013).
- [127] M. J. Hampden-Smith and T. T. Kodas, Chemical Vapor Deposition 1 (1995).
- [128] B. Mercey, P. A. Salvador, W. Prellier, T.-D. Doan, J. Wolfman, J.-F. Hamet, M. Hervieu, and B. Raveau, Journal of Materials Chemistry 9, 233 (1999).
- [129] D. B. Mitzi, Chemical Materials 13, 3283 (2001).
- [130] D. M. Deaven and K. M. Ho, Physical Review Letters **75**, 288 (1995), arXiv:9506004 [mtrl-th].
- [131] S. M. Woodley, P. D. Battle, J. D. Gale, and C. Richard A. Catlow, Physical Chemistry Chemical Physics 1, 2535 (1999).
- [132] B. G. Sumpter and D. W. Noid, Annual Review of Materials Science 26, 223 (1996).
- [133] S. Curtarolo, D. Morgan, and G. Ceder, Calphad: Computer Coupling of Phase Diagrams and Thermochemistry 29, 163 (2005), arXiv:0502465 [cond-mat].

- [134] A. N. Kolmogorov and S. Curtarolo, Physical Review B Condensed Matter and Materials Physics **73**, 1 (2006), arXiv:0603304 [cond-mat].
- [135] K. Kang, Y. S. Meng, J. Breger, C. P. Grey, and G. Ceder, Science **311**, 977 (2006).
- [136] H. Chen, G. Hautier, A. Jain, C. Moore, B. Kang, R. Doe, L. Wu, Y. Zhu, Y. Tang, and G. Ceder, Chemistry of Materials 24, 2009 (2012).
- [137] G. Hautier, A. Jain, T. Mueller, C. Moore, S. P. Ong, and G. Ceder, Chemistry of Materials 25, 2064 (2013).
- [138] J. W. Bennett, K. F. Garrity, K. M. Rabe, and D. Vanderbilt, Physical Review Letters 109, 1 (2012), arXiv:1206.4732v1.
- [139] C. V. Ciobanu, D. T. Tambe, and V. B. Shenoy, Surface Science 582, 145 (2005).
- [140] S. Wang, Z. Wang, W. Setyawan, N. Mingo, and S. Curtarolo, Physical Review X 1, 1 (2011).
- [141] J. Yan, P. Gorai, B. Ortiz, S. Miller, S. A. Barnett, T. Mason, V. Stevanovic, and E. S. Toberer, Energy & Environmental Science 8, 983 (2015).
- [142] D. Morgan, G. Ceder, and S. Curtarolo, Measurement Science and Technology 16, 296 (2005).
- [143] J. Behler, International Journal of Quantum Chemistry 115, 1032 (2015).
- [144] L. M. Ghiringhelli, J. Vybiral, S. V. Levechenko, C. Draxl, and M. Scheffler, Physical Review Letters 114 (2015).
- [145] (2019).
- [146] .
- [147] A. Jain, G. Hautier, C. J. Moore, S. Ping Ong, C. C. Fischer, T. Mueller, K. A. Persson, and G. Ceder, Computational Materials Science 50, 2295 (2011).
- [148] S. Curtarolo, W. Setyawan, G. L. W. Hart, M. Jahnatek, R. V. Chepulskii, R. H. Taylor, S. Wang, J. Xue, K. Yang, O. Levy, M. J. Mehl, H. T. Stokes, D. O. Demchenko, and D. Morgan, Computational Materials Science 58, 218 (2012), arXiv:1308.5715.
- [149] J. Hachmann, R. Olivares-Amaya, S. Atahan-Evrenk, C. Amador-Bedolla, R. S. Sánchez-Carrera, A. Gold-Parker, L. Vogt, A. M. Brockway, and A. Aspuru-Guzik, Journal of Physical Chemistry Letters 2 (2011).
- [150] "National institute of materials science," (2017).
- [151] J. E. Saal, S. K. nd Murathan Aykol, B. MEredig, and C. Wolverton, JOM 65 (2013).

- [152] I. Foster, R. Ananthakrisnan, B. Blaiszik, K. Chard, R. Osborn, S. Tuecke, M. Wilde, and J. Wozniak, Big Data and HPC (2015).
- [153] S. R. Kalidindi, D. B. Brough, S. Li, A. Cecen, A. L. Blekh, F. Y. P. Congo, and C. Campbell, Materials Research Society Bulletin 41 (2016).
- [154] R. Ramprasad, R. Batra, G. Pilania, A. Mannodi-Kanakkithodi, and C. Kim, Nature Computational Materials 3 (2017).
- [155] J. Gubernatis and T. Lookman, Physical Review Materials 2 (2018).
- [156] N. Wagner and J. Rondinelli, Frontiers in Materials 3 (2016).
- [157] J. J. De Pablo, B. Jones, C. L. Kovacs, V. Ozolins, and A. P. Ramirez, Current Opinion in Solid State and Materials Science 18, 99 (2014), arXiv:arXiv:1011.1669v3.
- [158] M. D. Wilkinson, M. Dumontier, I. J. Aalbersberg, G. Appleton, M. Axton, A. Baak, N. Blomber, J.-W. Boiten, L. B. da Silva Santos, P. E. Bourne, J. Bouwman, A. J. Brookes, T. Clark, M. Crosas, I. Dillo, O. Dumon, S. Edmunds, C. T. Evelo, R. Finkers, A. Gonzalez-Beltran, A. J. Gray, P. Groth, C. Goble, J. S. Grethe, J. Heringa, P. A. 't Hoen, R. Hooft, T. Kuhn, R. Kok, J. Kok, S. J. Lusher, M. E. Martone, A. Mons, A. L. Packer, B. Persson, P. Rocca-Serra, M. Roos, R. van Schaik, S.-A. Sansone, E. Schultes, T. Sengstag, T. Slater, G. Strawn, M. A. Swertz, M. Thompson, J. van der Lei, E. van Mulligen, J. Velterop, A. Waagmeester, P. Wittenburg, K. Wolstencroft, J. Zhao, and B. Mons, Scientific Data 3 (2016).
- [159] M. de Jong, W. Chen, H. Geerlings, M. Asta, and K. A. Persson, Scientific Data 2, 150053 (2015).
- [160] E. Kim, K. Huang, A. Tomala, S. Matthews, E. Strubell, A. Saunders, A. McCallum, and E. Olivetti, Nature Scientific Data 4 (2017).