

AE 737 - MECHANICS OF DAMAGE TOLERANCE

LECTURE 18

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SCHEDULE

- 31 Mar - Strain based fatigue, project abstract due
- 5 Apr - Crack Growth, Homework 7 due, Homework 8 assigned
- 7 Apr - Crack Growth, Stress Spectrum
- 12 Apr - Retardation, Boeing Commercial Method
- 14 Apr - Exam Review, Homework 8 Due
- 19 Apr - Exam 2
- 21 Apr - Exam Solutions, Damage Tolerance

1. strain based fatigue
2. general trends
3. other factors affecting fatigue
4. mean stress effects
5. multiaxial loading

STRAIN BASED FATIGUE

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- The strain-based method gives greater detail, and validity at lower cycles
- It is still valid for high cycle fatigue
- Does not include crack growth analysis or fracture mechanics

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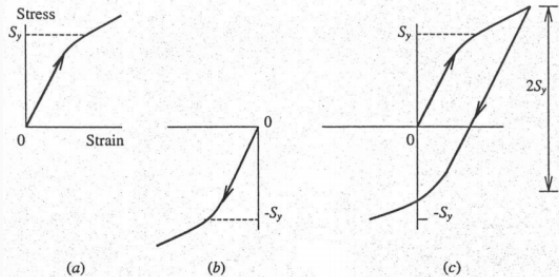
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- This is most commonly done using axial test machines (instead of rotating bending tests)
- The test is run in strain control (not load control)
- Generally plotted on log-log scale

- We can separate the total strain into elastic and plastic components

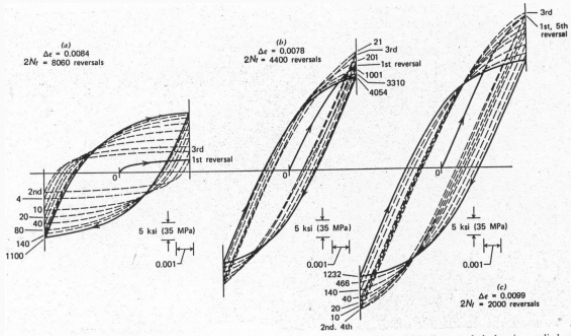
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$$\epsilon_a = \epsilon_{ea} + \epsilon_{pa} \quad (18.1)$$

PLASTIC STRAIN



HYSTERESIS LOOPS



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$$\epsilon_a = \frac{\sigma_a}{E} + \left(\frac{\sigma_a}{H'} \right)^{\frac{1}{n'}} \quad (18.2)$$

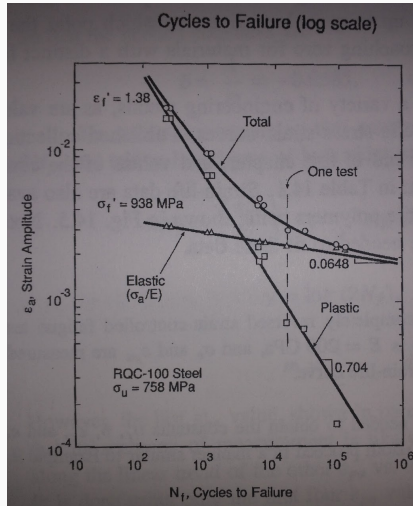
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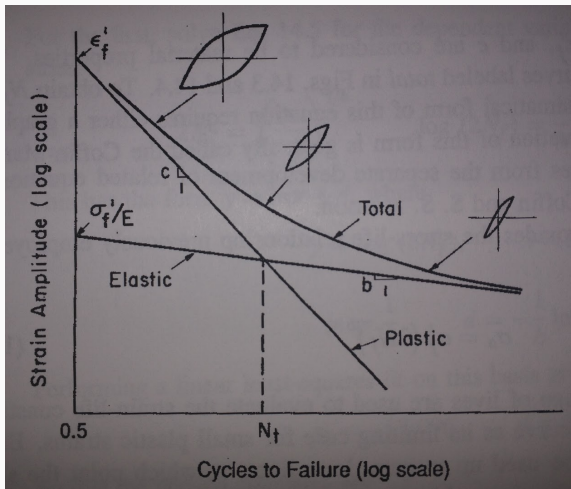
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- Once for total strain, once for plastic strain, and once for elastic strain
- Since plastic strain and elastic strain vary by the number of cycles, a hysteresis loop from half the fatigue life is generally used
- This is considered representative of stable behavior

EXPERIMENTAL DATA





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$$\epsilon_{ea} = \frac{\sigma'_f}{E} (2N_f)^b \quad (18.4)$$

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$$\epsilon_a = \frac{\sigma'_f}{E} (2N_f)^b + \epsilon'_f (2N_f)^c \quad (18.6)$$

Data from p. 270

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$$N_t = \frac{1}{2} \left(\frac{\sigma'_f}{\epsilon'_f} \right)^{\frac{1}{c-b}} \quad (18.7)$$

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- We can consider the plastic portion and solve for σ_a

$$\sigma_a = H' \epsilon_{pa}^{n'} \quad (18.9)$$

- We can eliminate $2N_f$ from the plastic strain equation

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- We then compare with 18.9 and find

$$H' = \frac{\sigma'_f}{(\epsilon'_f)^{b/c}} \quad (18.12a)$$

$$n' = \frac{b}{c} \quad (18.12b)$$

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- However, in practice these constants are fit from different curves
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- One cause for this is data that do not lie on a straight line in the log-log domain
- For ductile materials at short lives, the true stresses and strains may differ significantly from engineering stress and strain

GENERAL TRENDS

- We can consider a tensile test as a fatigue test with $N_f = 0.5$

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- And similarly for strain $\tilde{\epsilon}_f \approx \epsilon'_f$

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- In turn this intersects with the elastic strain line much later, resulting a longer transition life for ductile materials

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- It is also noteworthy that strain-life for many metals pass through the point $\epsilon_a = 0.01$ and $N_f = 1000$ cycles
- Steels also follow a trend with Brinell Hardness, the higher they are on the HB scale, the lower their transition life

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- While "shallow" slopes are around $b = -0.05$, common for hardened metals

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- At a sufficiently elevated temperature for any material

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- Strain life curves are not very sensitive to surface finish or residual strength at short lives
- The plastic deformation tends to remove residual stresses
- In high-cycle fatigue, crack initiation is important (poor surface finish allows cracks to form earlier)
- When plastic deformation is present (low-cycle fatigue), cracks form relatively quickly regardless of surface finish

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$$b_s = \frac{\log(m_s(2N_e)^b)}{\log(2N_e)} \quad (18.13)$$

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 - Hot rolling can also cause decarburization

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 - Shot peening introduces many small divots on surface, which can be detrimental in corrosion, but it does cause a residual compressive stress on the surface

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$$m_d = \left(\frac{d}{25.4\text{mm}} \right)^{-0.093} \quad (18.14)$$

- Which is then used to re-calculate material constants

$$\sigma'_{fd} = m_d \sigma'_f, \quad \epsilon'_{fd} = m_d \epsilon'_f \quad (18.15)$$

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- If the temperature is significantly different between two sides of a part thermal stresses can also be introduced
- Low temperatures generally cause a material to behave in a more brittle fashion, which alters the fatigue life
- High temperatures cause problems with creep-relaxation and can also affect the crystalline structure

MEAN STRESS EFFECTS

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- When the plastic strain is not significant, mean stress will exist
- Mean strain does not generally affect fatigue life

- Recall the Morrow approach for mean stress effects from the stress-based method

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- We can rearrange the equation such that

$$\sigma_a = \sigma'_f \left[\left(1 - \frac{\sigma_m}{\sigma'_f} \right)^{\frac{1}{b}} (2N_f) \right]^b \quad (18.17)$$

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$$N^* = N_f \left(1 - \frac{\sigma_m}{\sigma'_f} \right)^{\frac{1}{b}} \quad (18.18)$$

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$$N^* = N_f \left(1 - \frac{\sigma_m}{\sigma'_f} \right)^{\frac{1}{b}} \quad (18.18)$$

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$$\epsilon_a = \frac{\sigma'_f}{E} \left(1 - \frac{\sigma_m}{\sigma'_f} \right) (2N_f)^b + \epsilon'_f \left(1 - \frac{\sigma_m}{\sigma'_f} \right)^{\frac{c}{b}} (2N_f)^c \quad (18.19)$$

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- From the zero-mean stress graph, find the point corresponding to your applied strain
- For a non zero mean stress, this point represents (ϵ_a, N^*) , we can now solve for N_f using 18.18

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$$\epsilon_a = \frac{\sigma'_f}{E} \left(1 - \frac{\sigma_f}{\sigma'_f} \right) (2N_f)^b + \epsilon'_f (2N_f)^c \quad (18.20)$$

- There is no convenient solution method for this form, and it generally must be solved numerically, or plotted with many families of σ_m

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$$\sigma_{max}\epsilon_a = \frac{(\sigma'_f)^2}{E}(2N_f)^{2b} + \sigma'_f\epsilon'_f(2N_f)^{b+c} \quad (18.21)$$

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$$\sigma_{max}\epsilon_a = \frac{(\sigma'_f)^2}{E} (2N_f)^{2b} + \sigma'_f \epsilon'_f (2N_f)^{b+c} \quad (18.21)$$

- This method can also be solved graphically if a plot of $\sigma_{max}\epsilon_a$ is made using zero-mean data. All we need to do is find the new $\sigma_{max}\epsilon_a$ point to find a new N_f

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- The modified Morrow method gives improved results in many materials
- The SWT approach is very good for general use, but is non-conservative with a compressive mean stress

MULTIAXIAL LOADING

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- We are currently only capable of handling proportional loads that are in-phase (i.e. have the same frequency)
- If we consider the principal directions where $\sigma_{2a} = \lambda\sigma_{1a}$, we find an expression for the strain-life as

$$\epsilon_{1a} = \frac{\frac{\sigma'_f}{E}(1 - \nu\lambda)(2N_f)^b + \epsilon'_f(1 - 0.5\lambda)(2N_f)^c}{\sqrt{1 - \lambda + \lambda^2}} \quad (18.22)$$

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 1. Pure planar shear: $\lambda = -1, T = 0$

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$$\bar{\epsilon}_a = \frac{\sigma'_f}{E} (2N_f)^b + 2^{1-T} \epsilon'_f (2N_f)^c \quad (18.24)$$