

# AE837

## Advanced Mechanics of Damage Tolerance

Dr. Nicholas Smith

Wichita State University, Department of Aerospace Engineering August  
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## upcoming schedule

- Aug 27 - Griffith Fracture
- Aug 29 - Griffith Fracture
- Sep 3 - Elastic Stress Field, Homework 1 Due
- Sep 5 - Elastic Stress Field

## outline

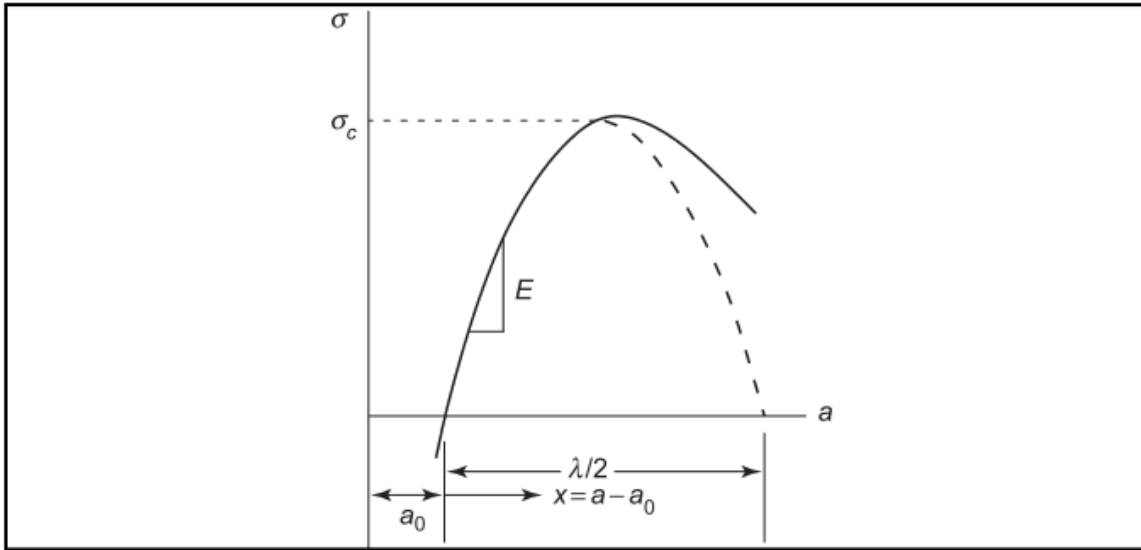
- theoretical strength
- griffith fracture
- energy release rate
- relation among energies

# theoretical strength

## theoretical strength

- We can (theoretically) predict the strength of a material from the strength of its atomic bonds
- As the distance between atoms increases (strain) so does the attractive force between them (stress) until some critical distance when they are no longer attracted to one another (failure)

# cohesive stress



## theoretical strength

- Using a sinusoidal approximation of strength we have

$$\sigma = \sigma_c \sin\left(\frac{2\pi x}{\lambda}\right)$$

- For small strains,  $\sin(x) \approx x$
- We can also write in terms of the modulus of elasticity,  $E = \frac{\sigma}{\epsilon} = \frac{\sigma}{x/a_0}$

## theoretical strength

- This means we can write  $\sigma = \frac{Ex}{a_0}$ , substituting into the original equation we find

$$\frac{Ex}{a_0} = \sigma_c \frac{2\pi x}{\lambda}$$

- solving for  $\sigma_c$ , we find the theoretical strength as

$$\sigma_c = \frac{\lambda E}{2\pi a_0}$$

- In many materials,  $\lambda \approx a_0$ , in which case  $\sigma_c = \frac{E}{2\pi}$



## energy consideration

- Surface energy is the work done to create a new surface when an atomic bond breaks
- We define  $\gamma$  as the surface energy (units of energy/area)
- We can calculate this surface energy in terms of the sinusoidal approximation of traction-separation

$$2\gamma = \int_0^{\lambda/2} \sigma_c \sin\left(\frac{2\pi x}{\lambda}\right) dx = \frac{\lambda\sigma_c}{\pi}$$

## energy consideration

- re-arranging a previous result, we know that

$$\lambda = \frac{2\pi a_0 \sigma_c}{E}$$

- from which we find

$$\sigma_c^2 = \frac{2\pi\gamma E}{2\pi a_0} = \frac{\gamma E}{a_0}$$

- For many materials, the surface energy is approximately  $\gamma = 0.01Ea_0$ , which gives an approximate theoretical strength of

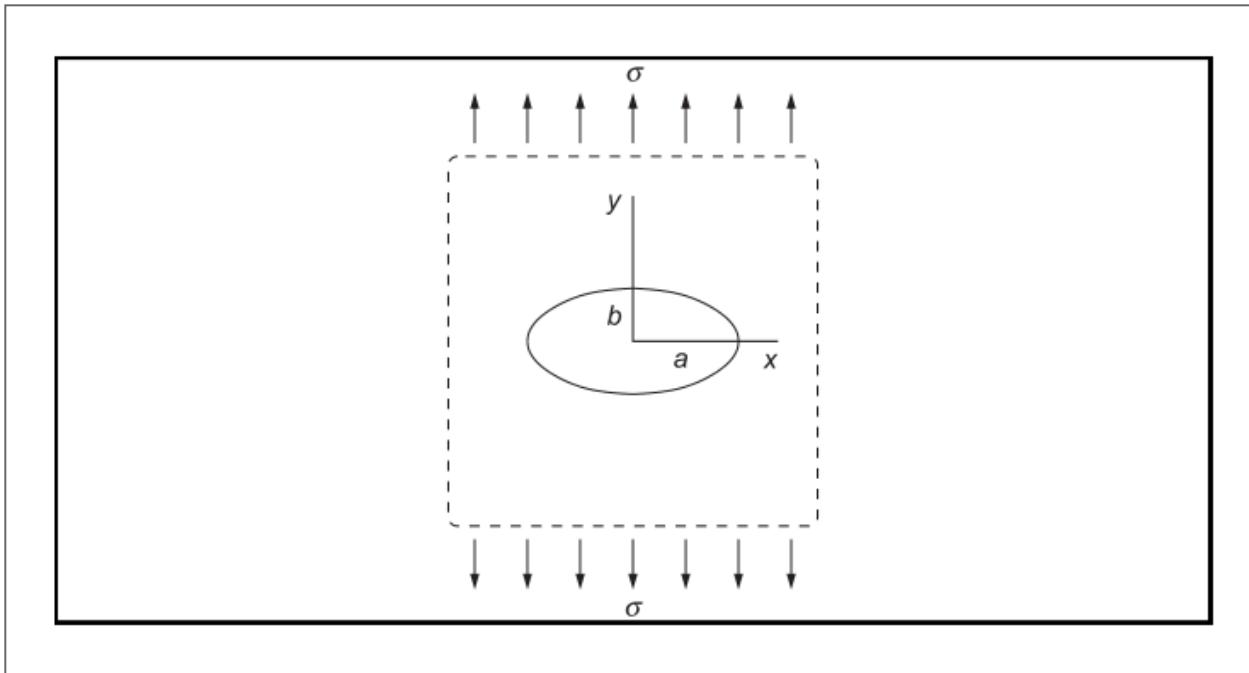
$$\sigma_c = \frac{E}{10}$$

# griffith fracture

## real failure

- In practice, materials fail at loads 10 to 100 times lower than the theoretical “atomic” strength
- Griffith studied glass, which by the atomic strength models should be a very strong material, but in practice is often weaker than much less stiff materials
- He hypothesized that there were many micro-cracks which coalesced to cause failure

# elliptic hole



## elliptic hole

- Inglis solved the problem of an elliptic hole under remote tension, finding that at the edge of the major axis the stress is given by

$$\sigma_{yy} = \sigma \left( 1 + \frac{2a}{b} \right)$$

- For a crack as  $b \rightarrow 0$ , we would get the unreasonable prediction that any stress, no matter how small, would produce failure as the stress at the edge of the ellipse would be infinite
- Griffith took an energy balance approach

## surface energy

- Griffith proposed that a crack would extend when the incremental release of energy,  $dW$  associated with a crack extension of  $da$  in a body is greater than the energy required to create the new surfaces,  $dW_s$
- For a center crack, there are two crack tips and a total of 4 surfaces, thus
$$W_s = 2(2a\gamma) = 4a\gamma$$

## surface energy

- Griffith then used the Inglis solution to calculate the total energy released by a crack extension of  $da$

$$W = \frac{\pi a^2 \sigma^2 (1 - \nu^2)}{E} \quad \text{plane strain}$$

$$W = \frac{\pi a^2 \sigma^2}{E} \quad \text{plane stress}$$



## surface energy

- We can now substitute to find under what conditions  $dW \geq dW_s$

$$\frac{2\pi a^2 \sigma_{cr}^2 (1 - \nu^2)}{E} da = 4\gamma da \quad \text{plane strain}$$

$$\frac{2\pi a^2 \sigma_{cr}^2}{E} da = 4\gamma da \quad \text{plane stress}$$

## surface energy

- Which gives the critical stress as

$$\sigma_{cr} = \sqrt{\frac{2E\gamma}{\pi(1 - \nu^2)a}} \quad \text{plane strain}$$

$$\sigma_{cr} = \sqrt{\frac{2E\gamma}{\pi a}} \quad \text{plane stress}$$

## comparison

- We can compare this critical stress prediction with previous methods
- We had found

$$\sigma_c = \sqrt{\frac{\gamma E}{a_0}}$$

- Thus when there is inherent damage in a material larger than the twice the characteristic distance between atoms ( $a_0$ ), the fracture strength will be lower than the theoretical strength

# energy release rate

## energy release rate

- We can now define the energy release rate
- More precisely, “strain energy release rate per crack tip”

$$dW = 2Gda$$

- Which after substitution and integration gives

$$G = \frac{1}{2} \frac{dW}{da} = \frac{\pi a \sigma^2 (1 - \nu^2)}{E} \quad \text{plane strain}$$
$$= \frac{\pi a \sigma^2}{E} \quad \text{plane stress}$$

## fracture toughness

- When  $G \geq 2\gamma$  the strain energy from the applied load is greater than the energy required to propagate a crack, and thus a crack will propagate
- The critical strain energy release rate is called  $G_c$ , and is also referred to as the fracture toughness

## fracture toughness

- Note:  $K_c$  (the critical stress intensity factor) is also referred to as the “fracture toughness,” and although they give equivalent predictions, they have different units
- The  $G_c$  predicted from the atomic surface energy will be somewhat conservative, since almost all materials will absorb at least some of the strain energy in plastic deformation

# relation among energies



## energy balance

- Griffith's theory is founded on the principle of energy balance
- During crack extension, the external work done,  $dW_e$  must equal the increment of surface energy,  $dW_s$ , and the increment of elastic strain energy,  $dU$

$$dW_s + dU = dW_e$$

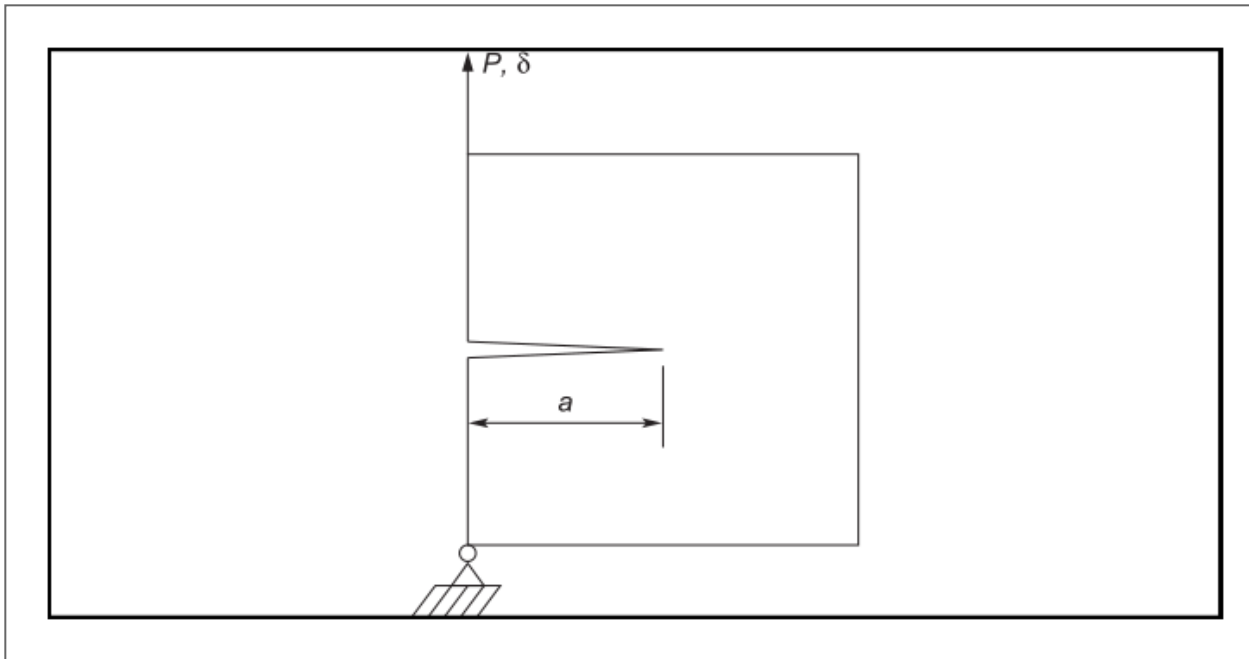
## energy balance

- For a conservative field we can write this as

$$\frac{\partial}{\partial a}(W_s + U + V) = 0$$

- Where a negative potential,  $V$ , implies positive external work done  $dW_e$

# example



## example

- The displacement at the top edge will be proportional to the applied load by some elastic compliance of the specimen

$$\delta = SP$$

- Note that this compliance,  $S$ , will be a function of the crack length,  $a$
- The strain energy can be expressed as

$$U = \int_{\delta=0}^{\delta=SP} P d\delta = \int_{\delta=0}^{\delta=SP} \frac{\delta}{S} d\delta$$

## example

- After integrating

$$U = \frac{1}{2S} (\delta^2) \Big|_0^{SP} = \frac{1}{2} SP^2$$

## example

- To find the incremental strain energy increase (where both  $P$  and  $S$  should be treated as variable), we find

$$dU = \frac{1}{2}P^2dS + SPdP$$

- We will now consider two loading cases, one with fixed displacement and the other with fixed loading force

## fixed displacement

- Under constant displacement we have  
 $\delta = SP = \text{constant}$
- This means that the derivative of displacement will be zero, hence  
 $d\delta = SdP + PdS = 0$
- and  
 $SdP = -PdS$

## fixed displacement

- Substituting into the previous equation gives

$$dU = -\frac{1}{2}P^2 dS$$

- Since  $d\delta = 0$ , the external work,  $dW_e = 0$  and we find

$$dW_s = -dU = \frac{1}{2}P^2 dS$$



## fixed load

- If instead of fixing displacement we fix the applied load we have  
 $dP = 0$
- Which gives  
$$dU = \frac{1}{2} P^2 dS$$
- The strain energy increases (while under fixed displacement it decreased)

## fixed load

- Further, we can find the external work done as

$$dW_e = Pd\delta = P^2dS$$

- And thus, from energy balance, we find the surface energy

$$dW_s = \frac{1}{2}P^2dS$$

- In this case the external work is equally split between strain energy and surface energy

## comparison

- In both cases, the energy released is  $dW = dW_e - dU$

$$dW = 0 - \left(-\frac{1}{2}P^2 dS\right) \quad \text{fixed displacement}$$

$$dW = P^2 dS - \frac{1}{2}P^2 dS \quad \text{fixed load}$$

- And we see that the energy released is independent of the load type

## example - double cantilever beam

