

Phase Stability & Material Design

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RADIOLOGICAL SCIENCES
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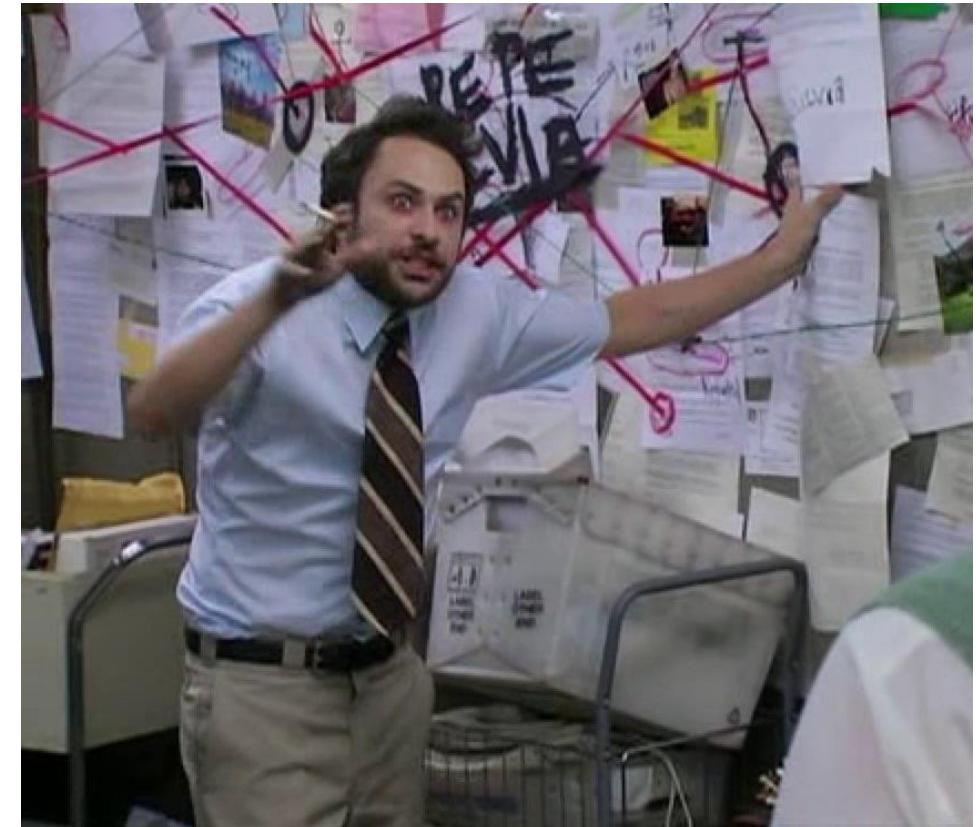
Don't forget to submit HW5!



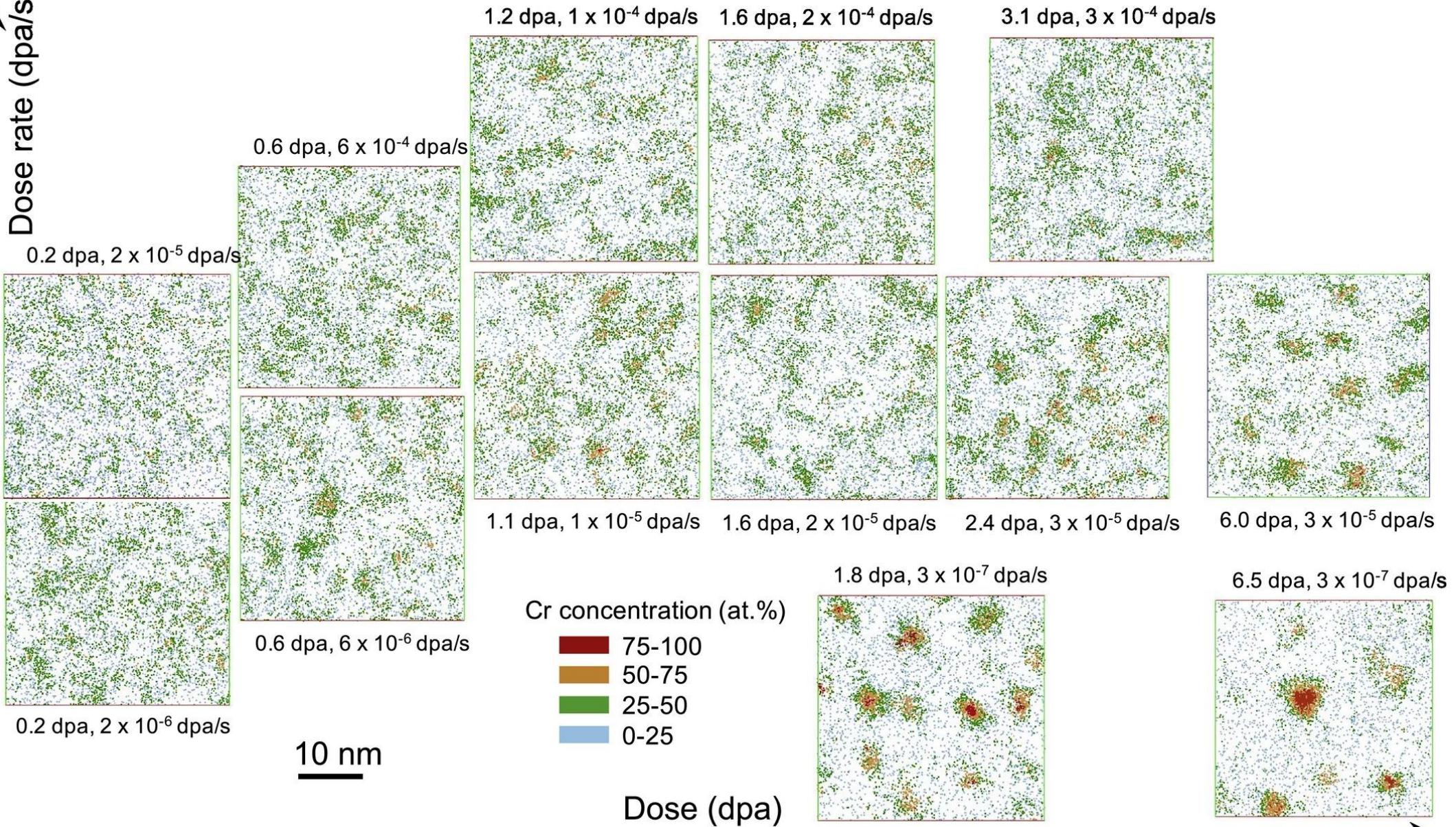
Final Exam Structure

- We will follow a similar format to the mid-term:
 - Non-cumulative: covers topics on previous slide

12/15 (Monday) from
10:30 am – 12:30 pm in
Cooley 2918



Dose rate (dpa/s)



Radiation-induced precipitate dissolution

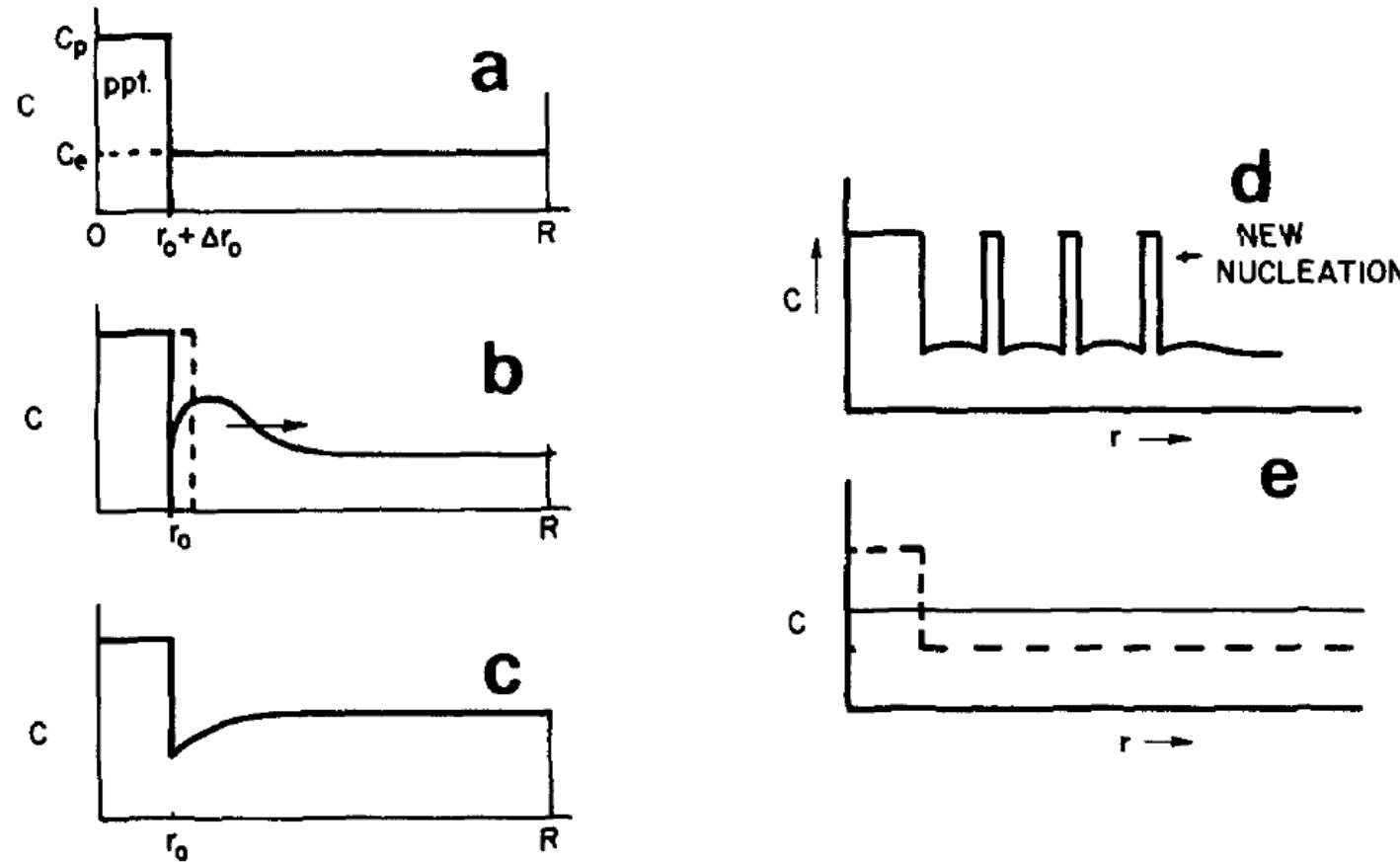


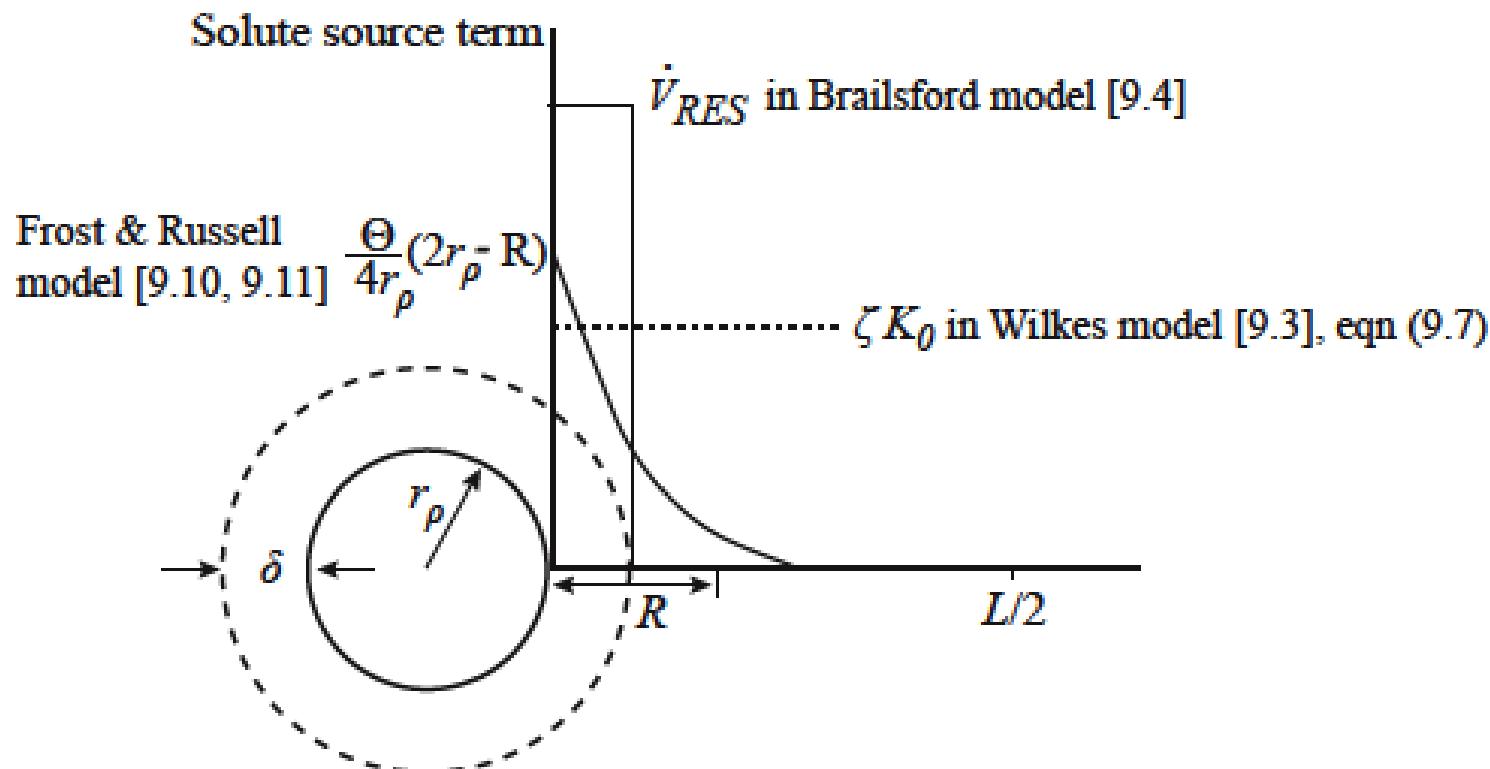
Fig. 3. A schematic diagram of concentration variations in a precipitate cell: (a) before irradiation with a precipitate of radius $r_0 + \Delta r_0$ in equilibrium with a uniform matrix concentration c_e ; (b) when irradiation begins surface dissolution causes a local solute supersaturation some of which diffuses outwards down the gradient; (c) at steady-state, the matrix is now uniformly supersaturated and the precipitate dissolution is matched by the back diffusion into the precipitate; (d) if the local supersaturation around the precipitate exceeds that required for nucleation, new precipitates may nucleate; (e) if the precipitates are widely spaced, the matrix may not reach the steady-state value and all precipitates will then dissolve.



Radiation-induced precipitate dissolution

In reality, modeling recoil distribution is inherently complex. Models that try to take this into account:

- Nelson model (NHM) – pg. 488-489 in Was
- Wilkes model – pg. 489-492 in Was
- Frost and Russell – pg. 492-493 in Was



Phase transformations without changes in composition

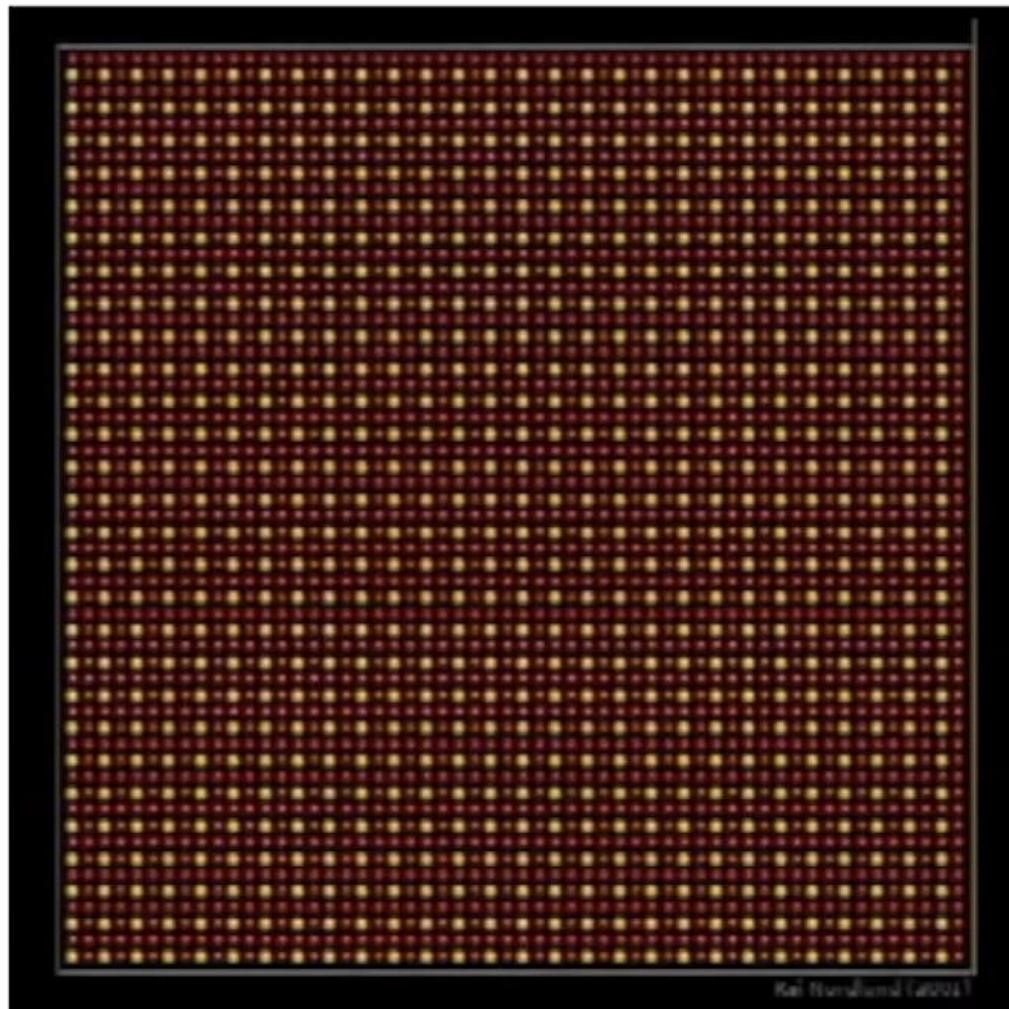


Radiation Disordering

9.1 Irradiation-induced disordering.

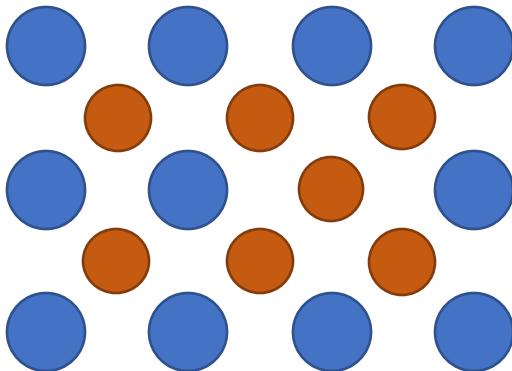
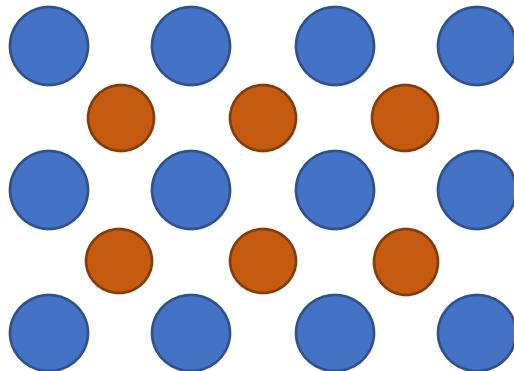
Disordering of Cu₃Au (L12 structure)
by bombardment with 10 keV Au ions.
(courtesy K. Nordlund, University of Helsinki)

Radiation disordering:
Loss of long-range
order



Radiation Disordering

- Antisite defects:
 - Occur in an ordered alloy or compound when atoms of different type exchange position



- We now need to consider random recombination so:



Radiation Disordering

- The steady state solution then becomes:

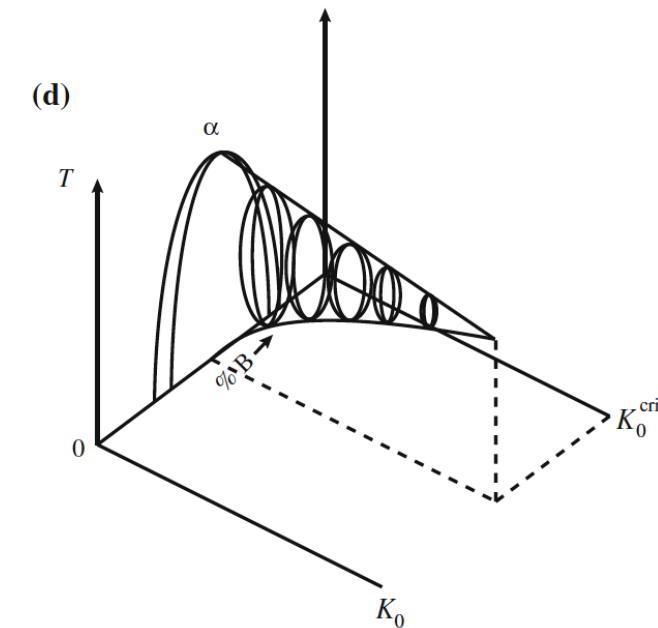
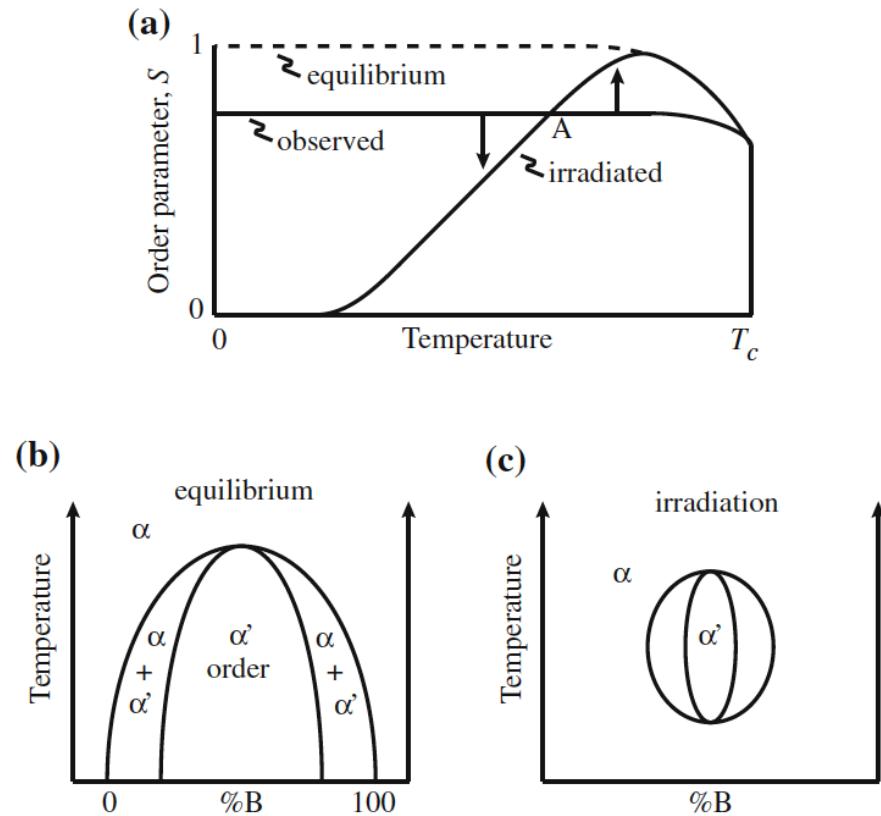
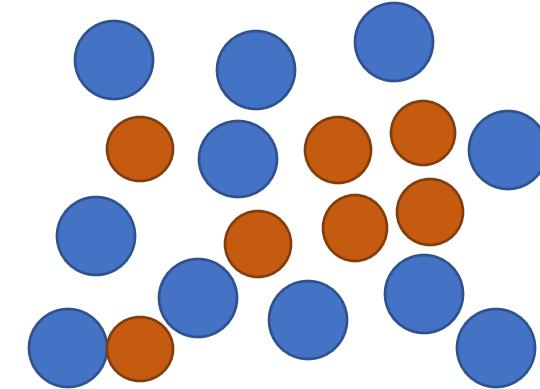
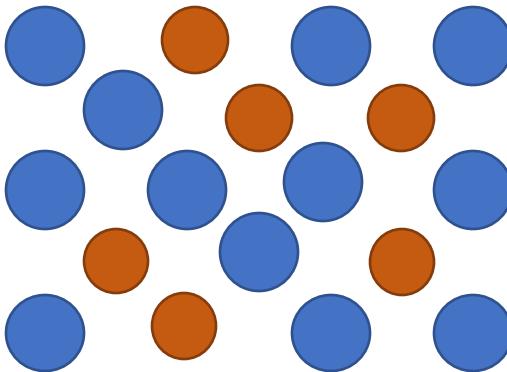
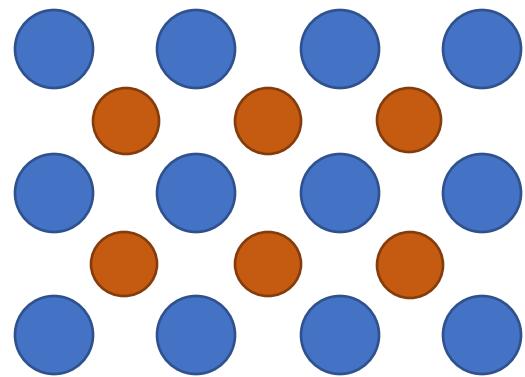


Fig. 9.11 (a) Effect of irradiation on the temperature dependence of the order parameter for a phase structure that is not at the equilibrium order, S_e . (b) Equilibrium phase diagram showing a first-order phase transformation of the disordered α phase to the ordered α' -phase. (c) Modification of the phase diagram by irradiation reflecting the change in order as given in (a). (d) Dependence on the irradiation-modified phase diagram with dose rate (after [3])



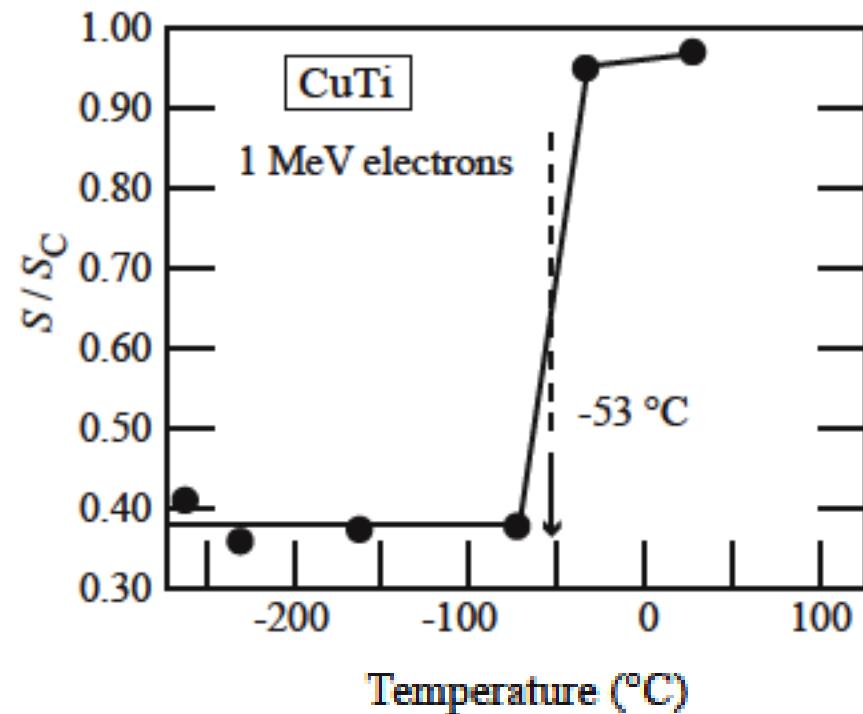
Temperature dependence of amorphization



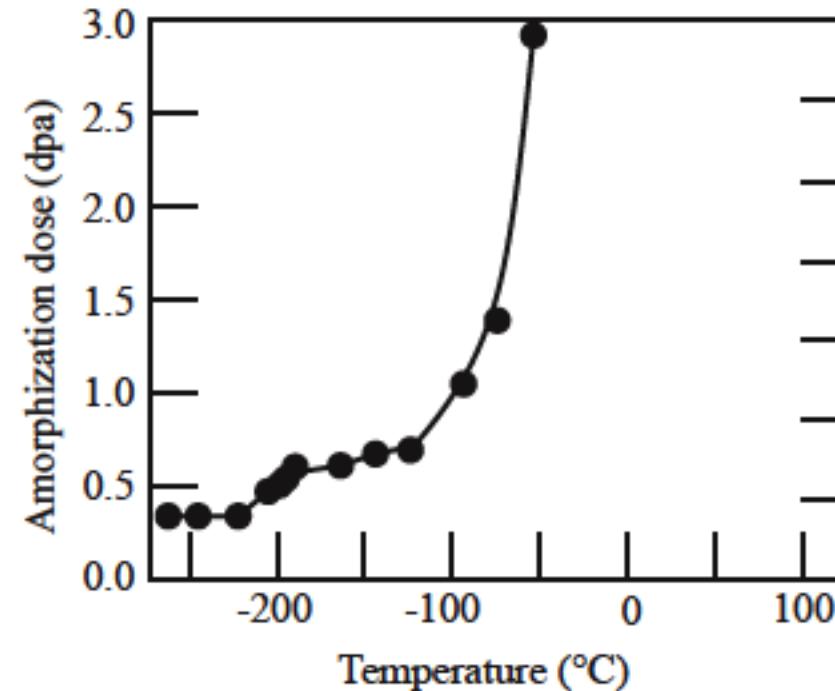
- Disorder means the destruction of the sub-lattice patterns in ordered alloys (e.g., creation of anti-site defects)
- Amorphization by particle irradiation refers to the complete loss of long-range crystalline structure



Temperature dependence of amorphization



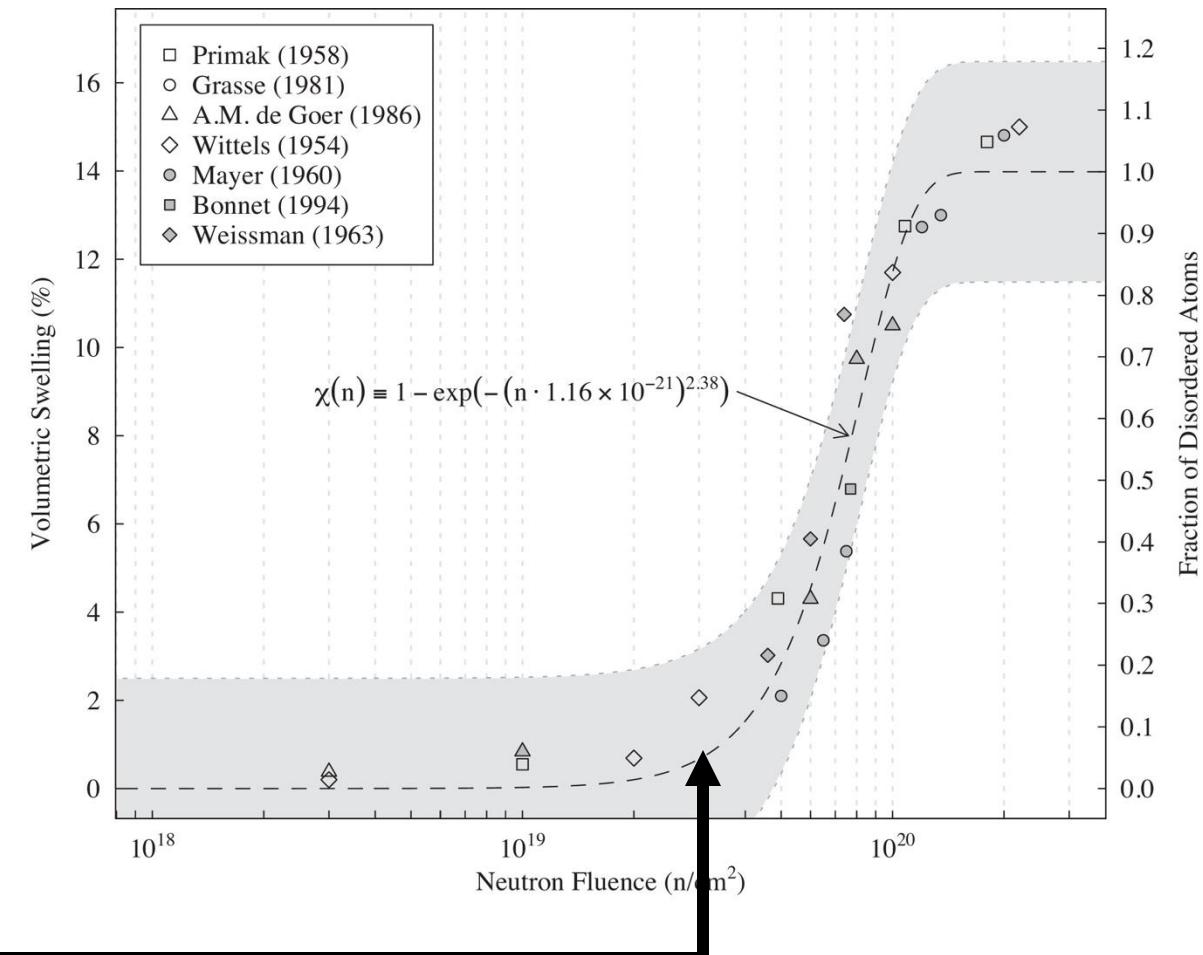
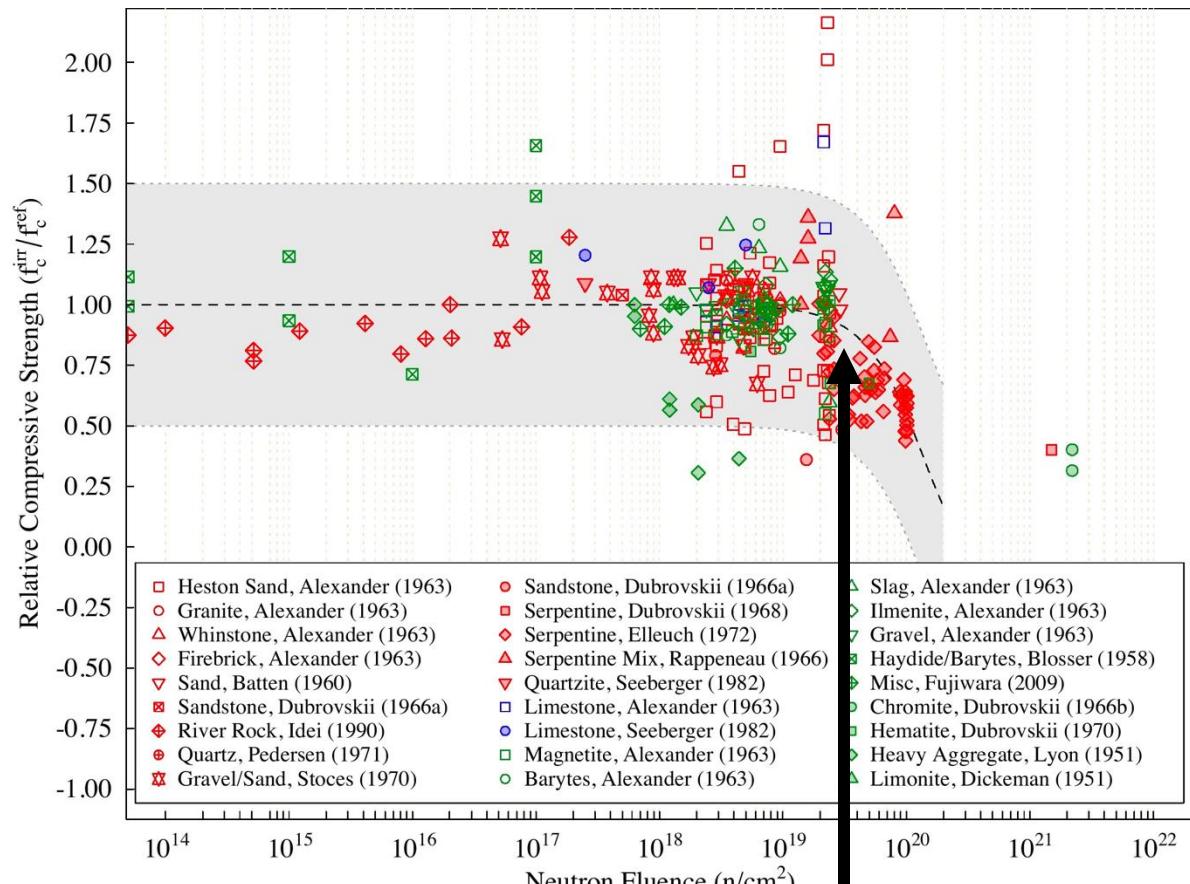
a



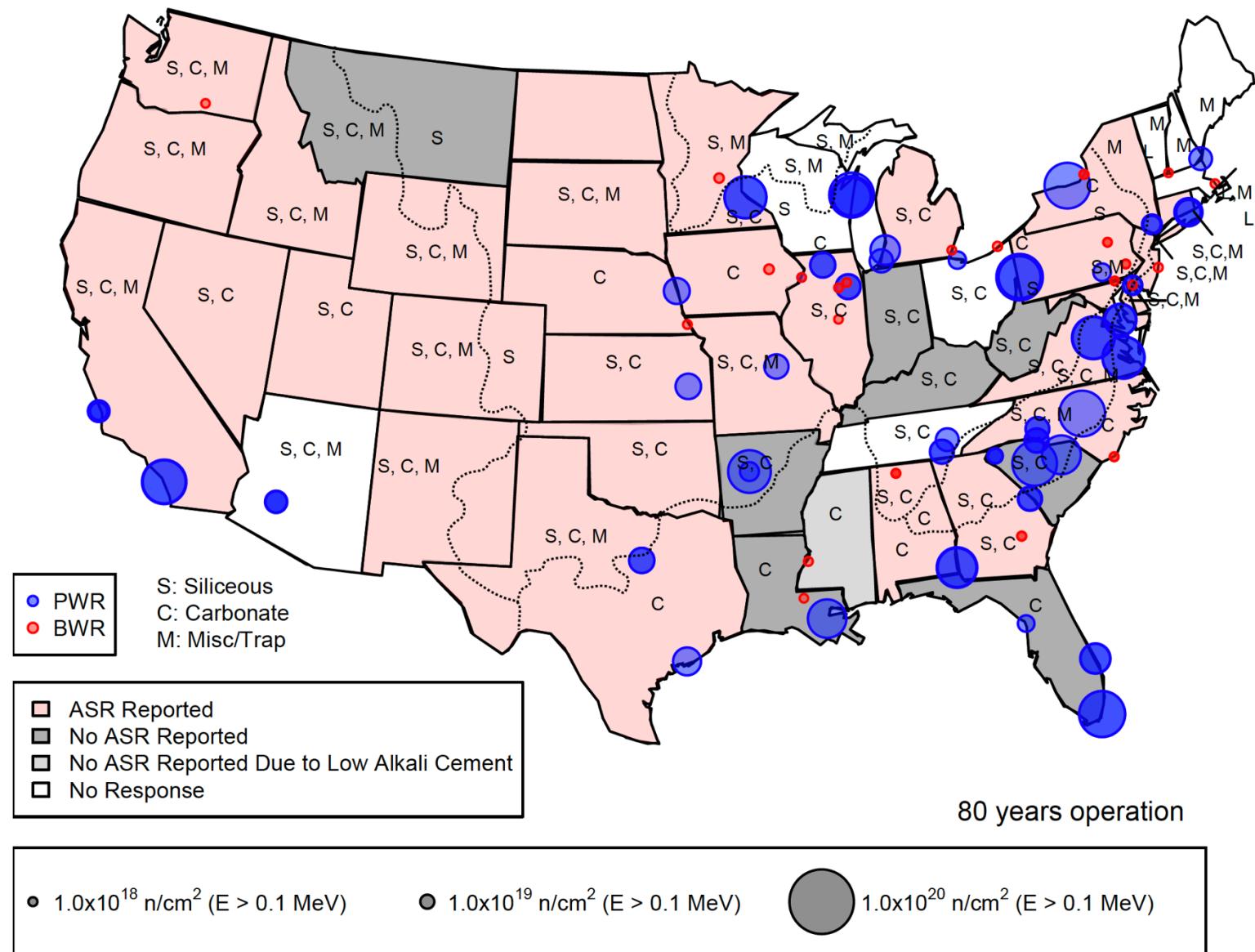
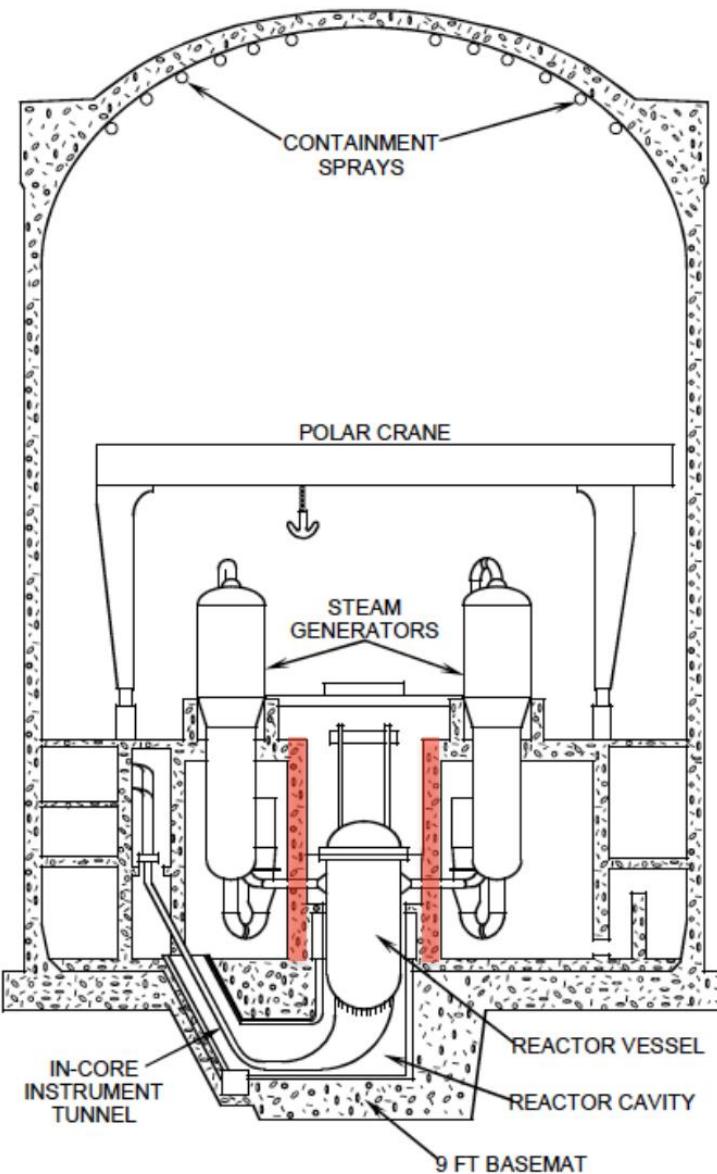
b

Temperature dependence of a) the long-range order parameter S/S_C , and b) the critical amorphization dose for CuTi, irradiated with 1 MeV electrons

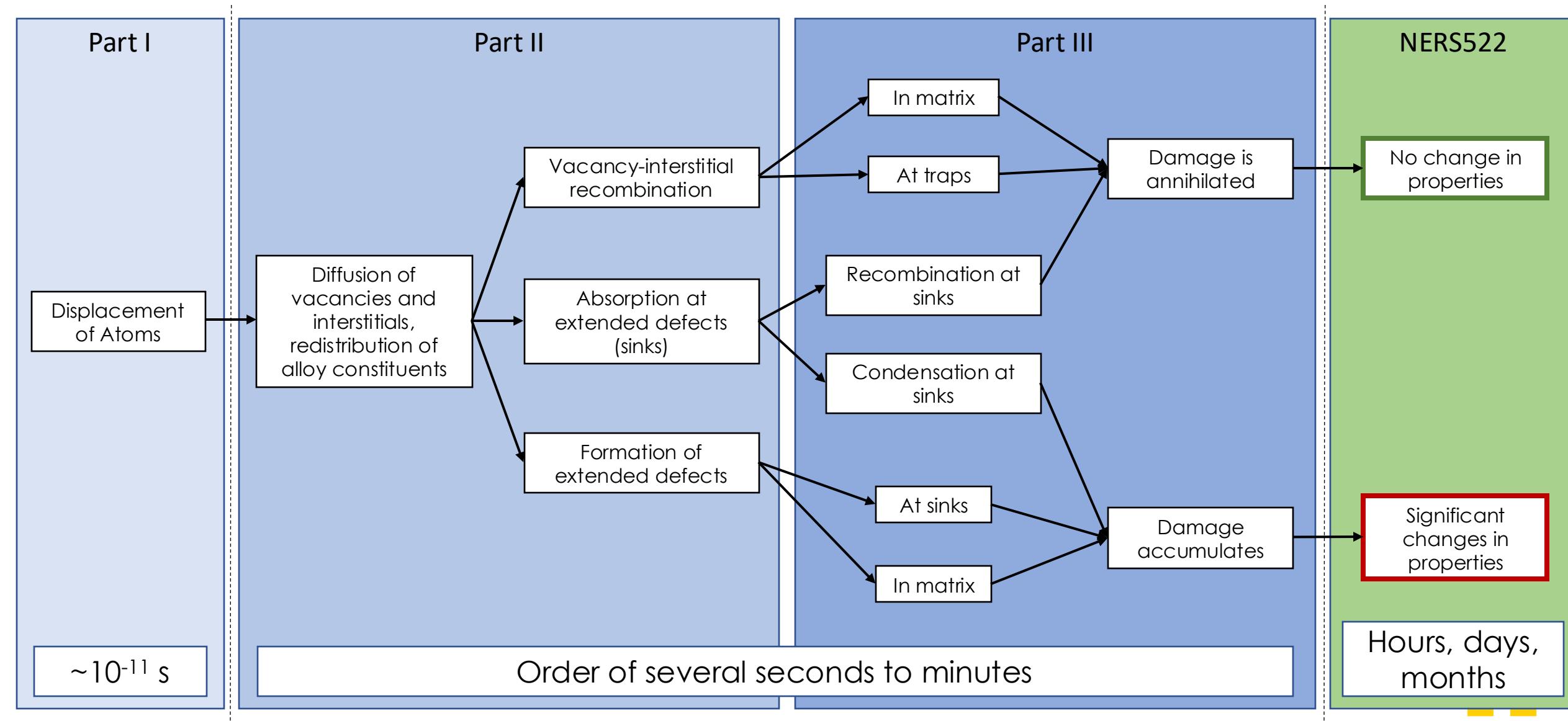
Irradiation-induced amorphization



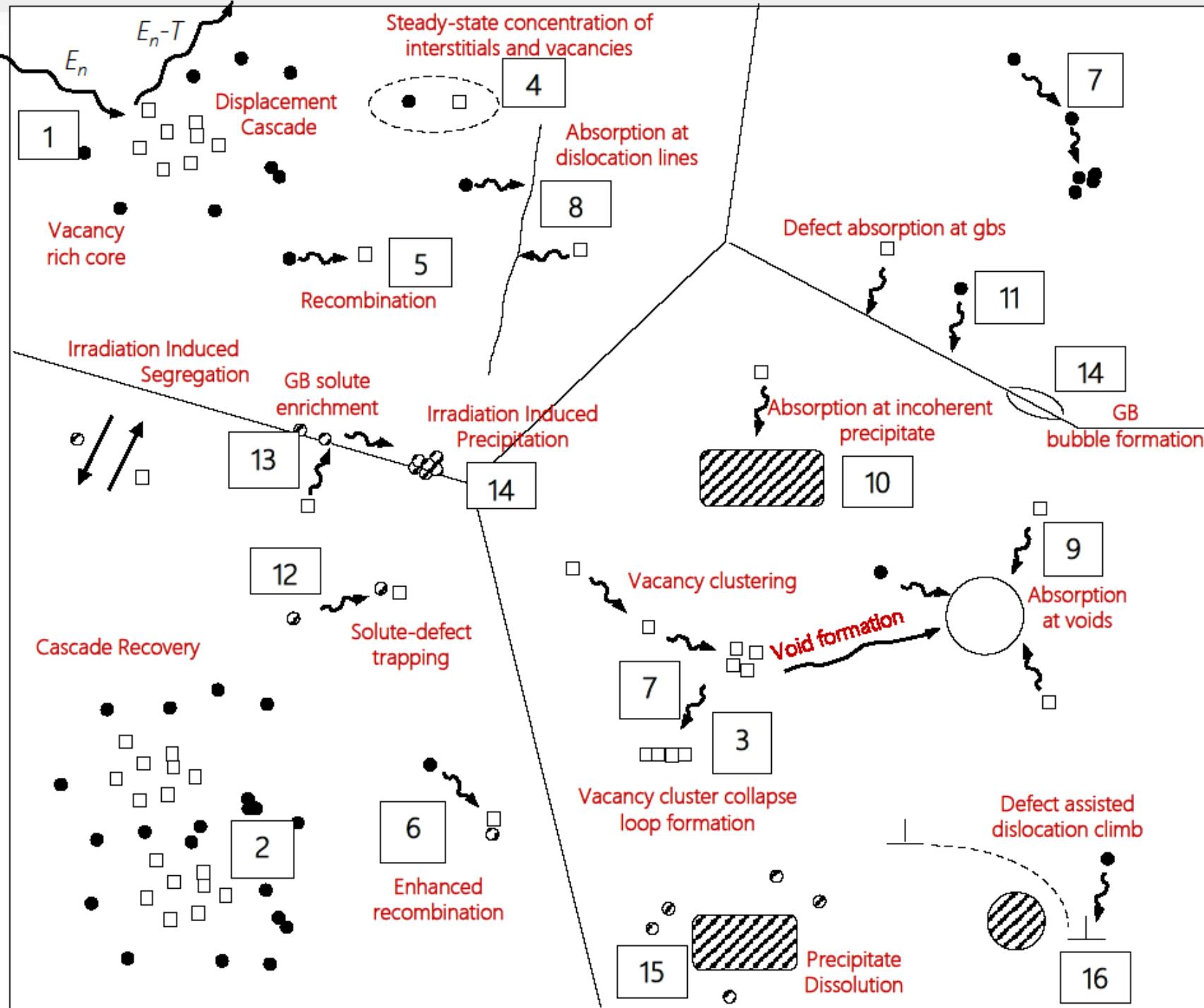
Irradiation-induced amorphization



Flow chart for radiation damage



Putting it into a visual



What is the month, day, and year of the earliest recorded snowflakes in the state of Michigan history?

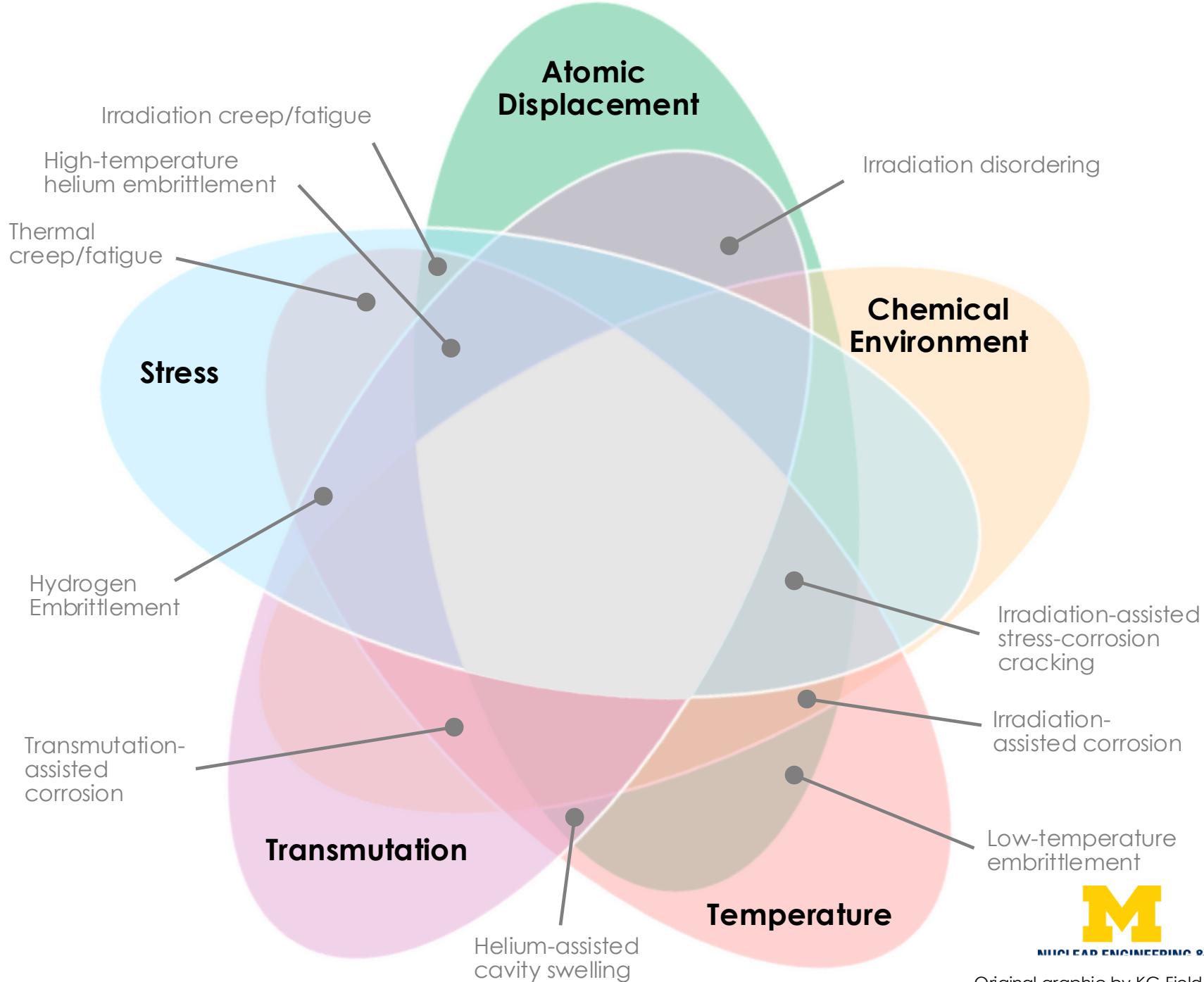


Applying the concepts of
NERS521 to material design



Materials degradation in a fusion and fission energy system can occur due to single- and **combined-effects**

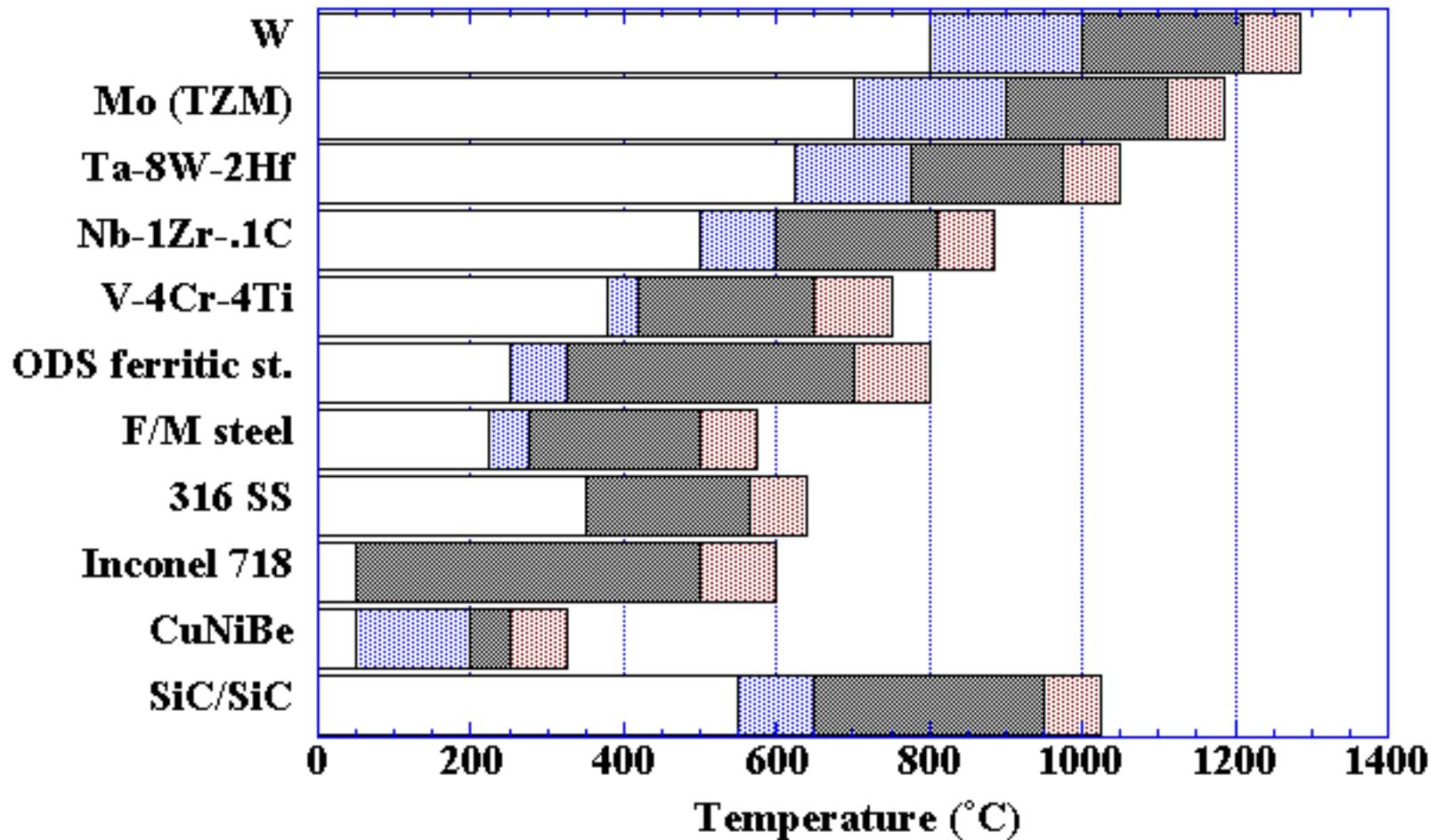
Listed degradation modes for structural materials



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Original graphic by KG Field

Bringing it together: Radiation Materials Design



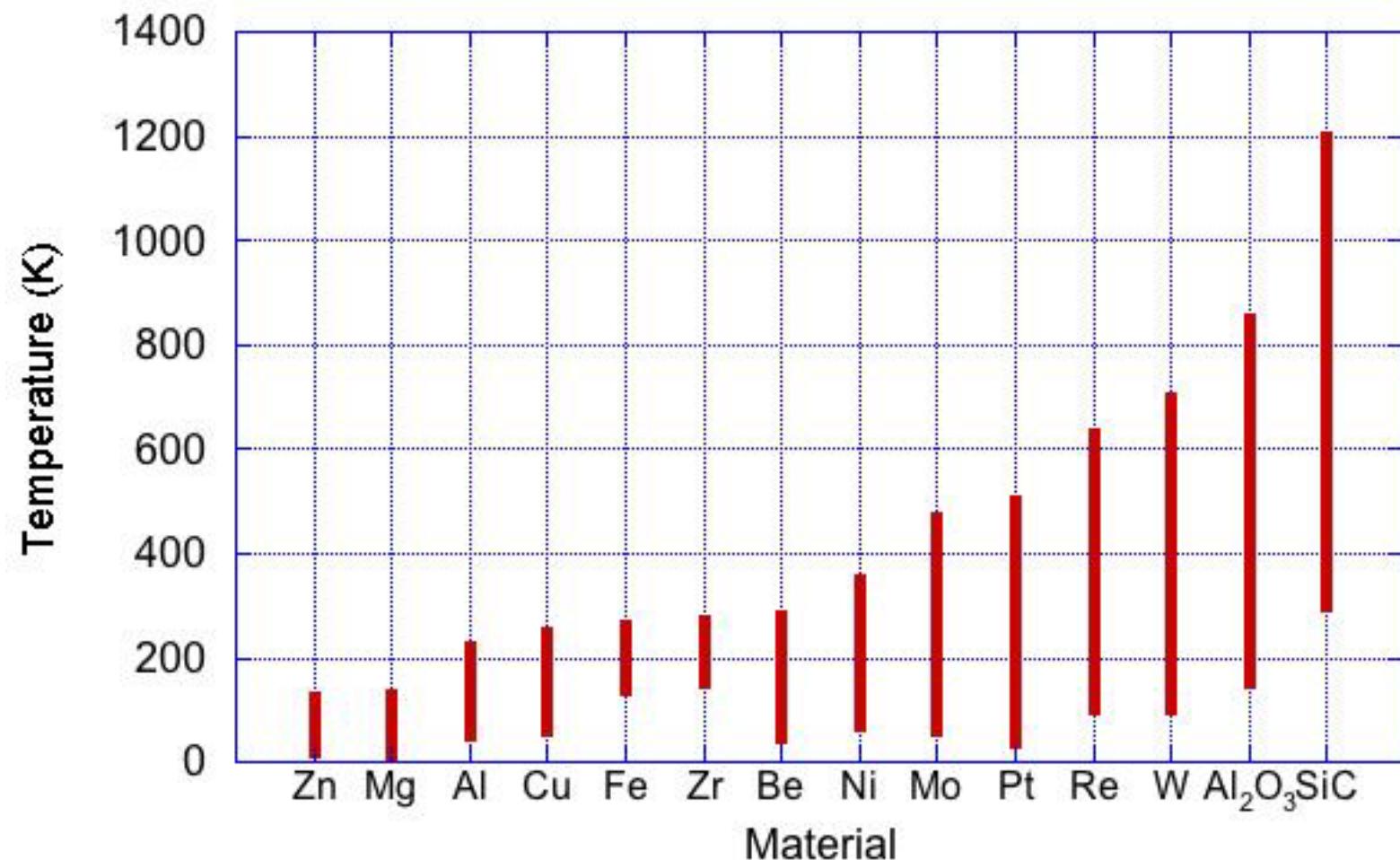
Zinkle and Ghoniem, *Fusion Engr. Des.* 49-50(2000) 709; S.J.
Zinkle & J.T. Busby, *Mater. Today* 12(2009) 12

Three options exist for designing radiation resistance materials

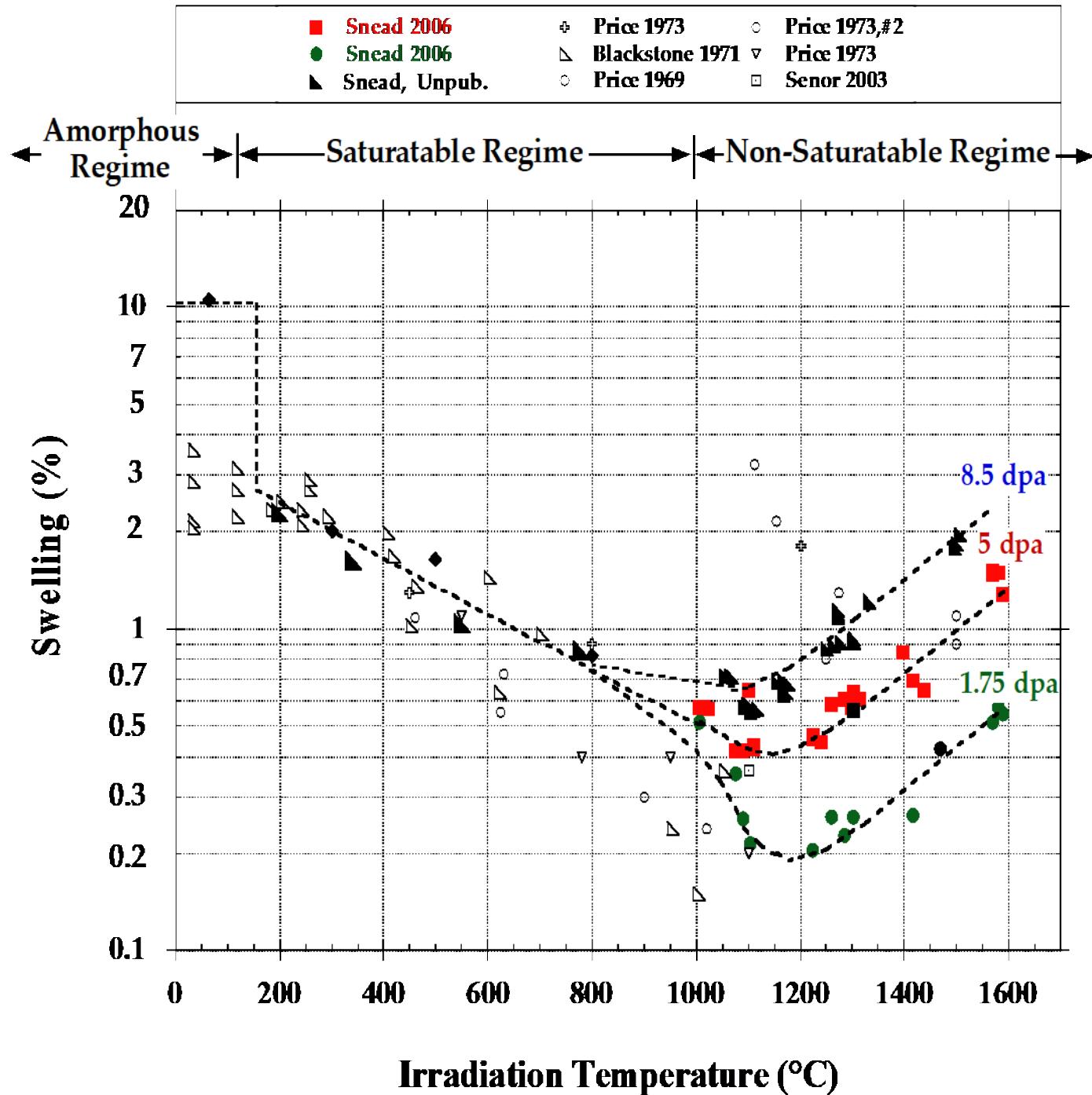
1. Use materials with negligible point defect mobility at desired operating temperatures
 - Slow down diffusion, but ideally want one defect type (e.g. vacancy/interstitial) mobile but the other not (to avoid amorphization/disordering)
2. Use materials with intrinsic resistance to radiation damage accumulation
 - Increased defect formation energies (e.g. stacking fault energies)
 - Examples: BCC alloys, high entropy/multicomponent alloys, nanocrystalline materials
3. Make materials with a high density of benign nanoscale recombination sites
 - Add microstructural features such as precipitates or nanolayered structures



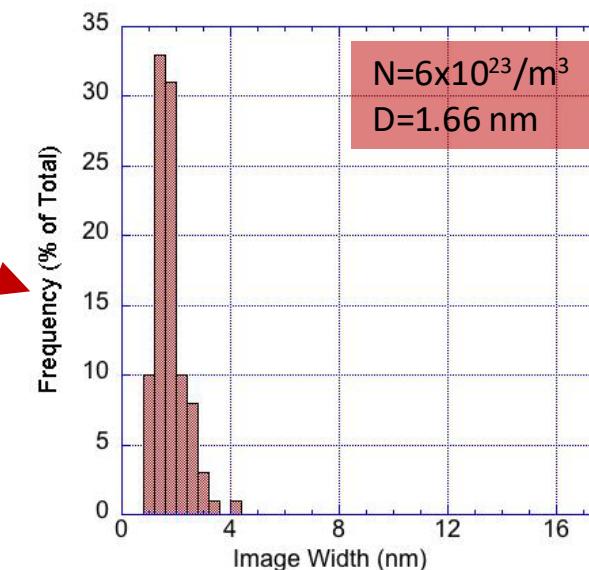
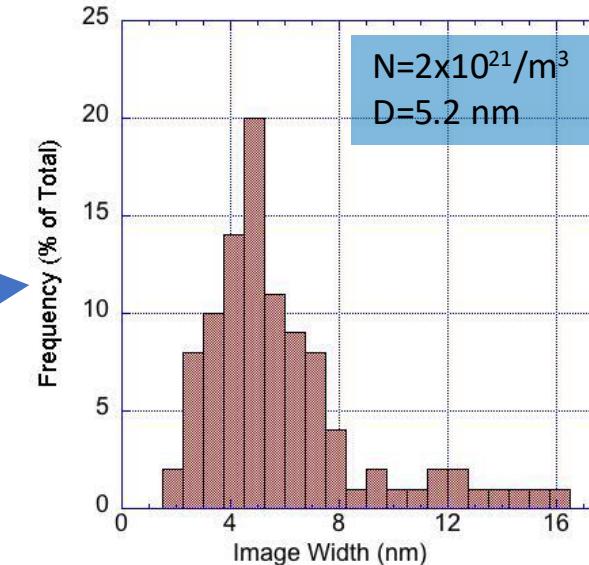
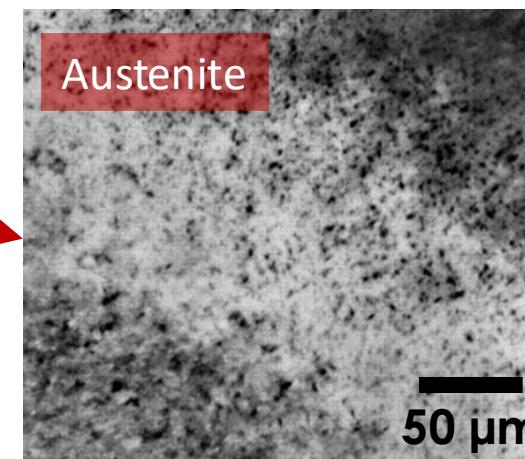
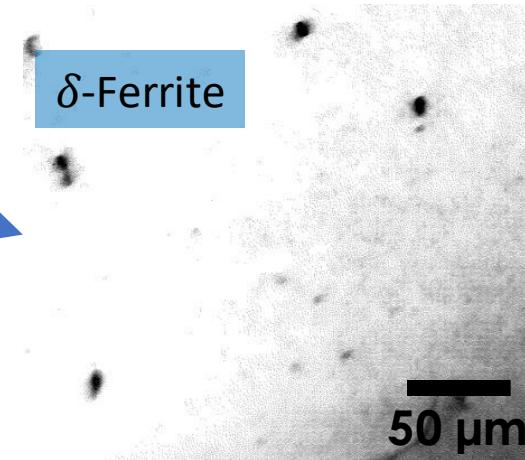
Design for Radiation Resistance I: Immobile point defects



Temperature regime where intrinsically high point defect recombination typically occurs



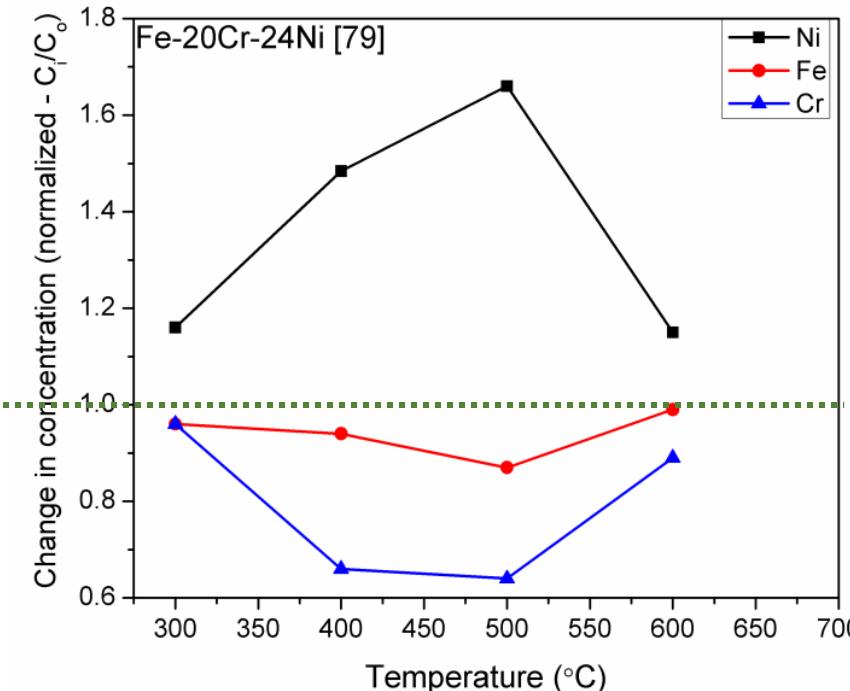
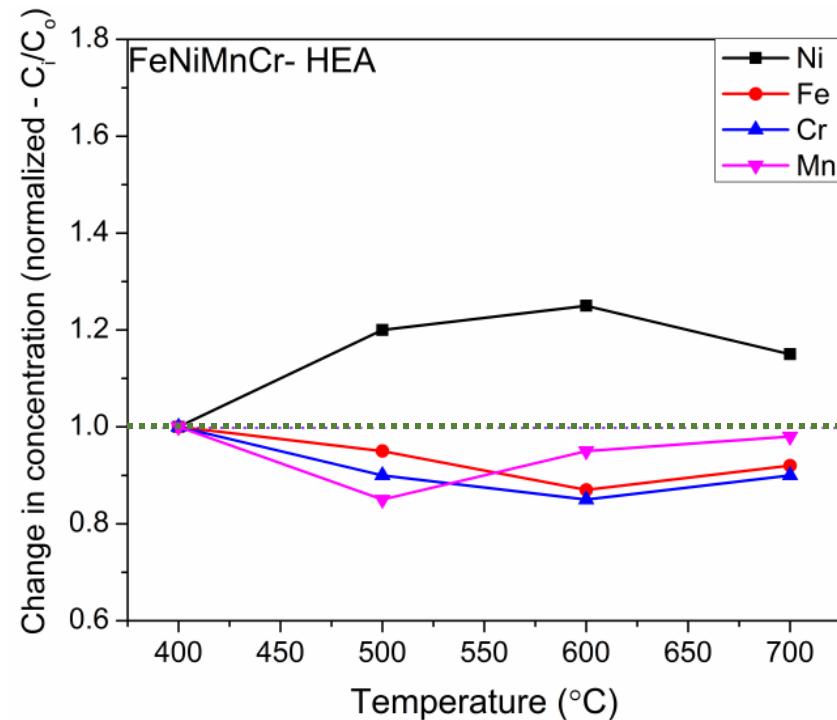
Design for Radiation Resistance II: Use radiation resistant phases



30x lower
visible defect
accumulation
for BCC phase



Design for Radiation Resistance II: Use radiation resistant phases

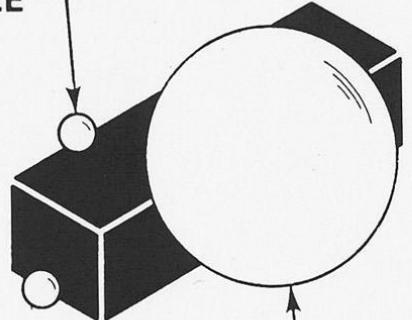


- RIS behavior for High Entropy Alloys is less pronounced than in traditional FCC FeCrNi(Mn) Alloys
=> HEA effect on solute diffusivities

Design for Radiation Resistance III: High Sink Strength

MICROSTRUCTURE OF LOW-SWELLING ALLOY TRAPS HELIUM IN MANY SUB-CRITICAL BUBBLES

CRITICAL-SIZE
BUBBLE



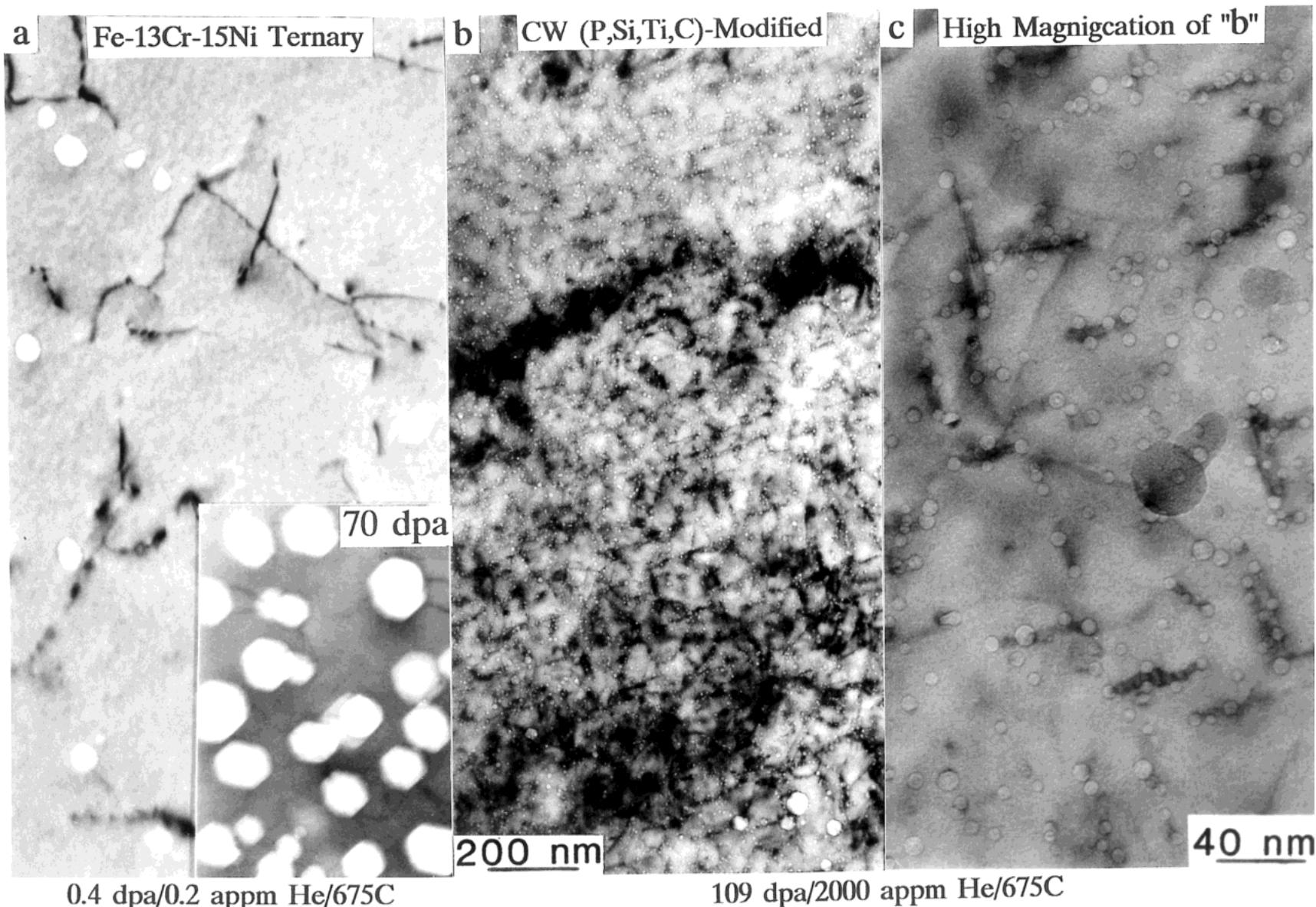
RAPIDLY GROWING
VOID

A FEW LARGE PARTICLES
(HIGH-SWELLING)



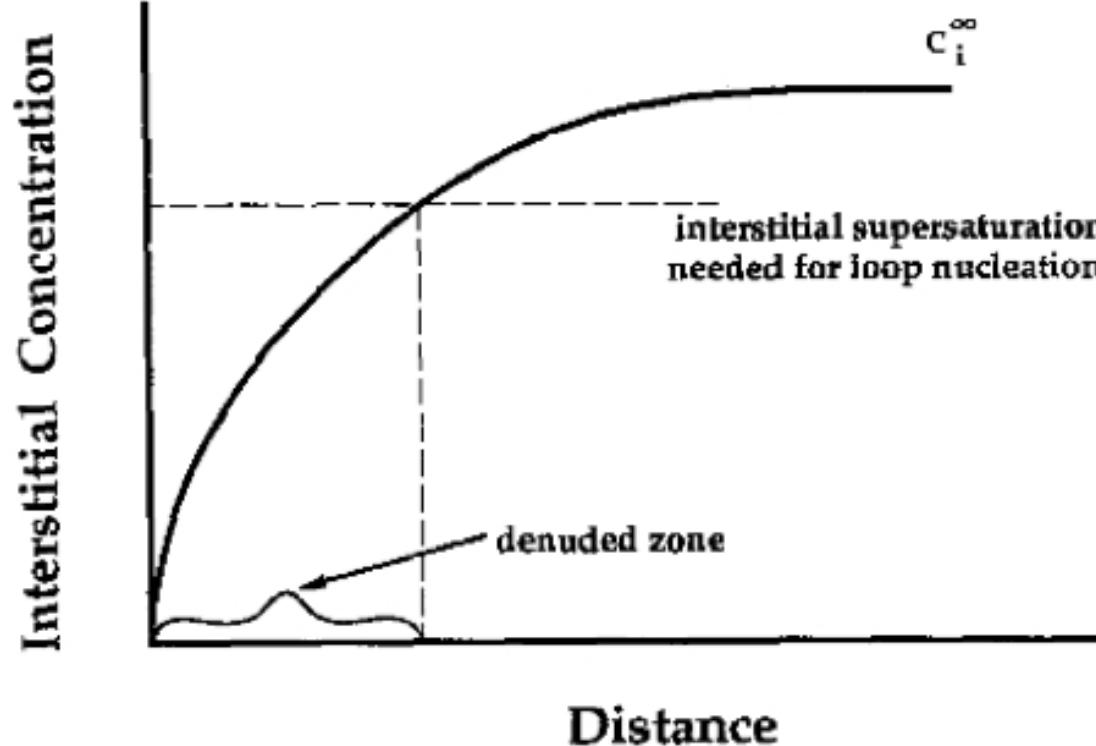
DISPERSED FINE PARTICLES
(LOW-SWELLING)

Design for Radiation Resistance III: High Sink Strength



Design for Radiation Resistance III: High Sink Strength

- Early research also investigated fine grained architectures for radiation resistance, but sufficiently stable nanoscale grain boundaries were not discovered
 - Fine-grained materials are susceptible to radiation-enhanced grain growth



S.J. Zinkle, Nucl. Instr. Meth. B 91(1994)234

L.K. Mansur & E.H. Lee, J. Nucl. Mater. 179-181(1991) 105

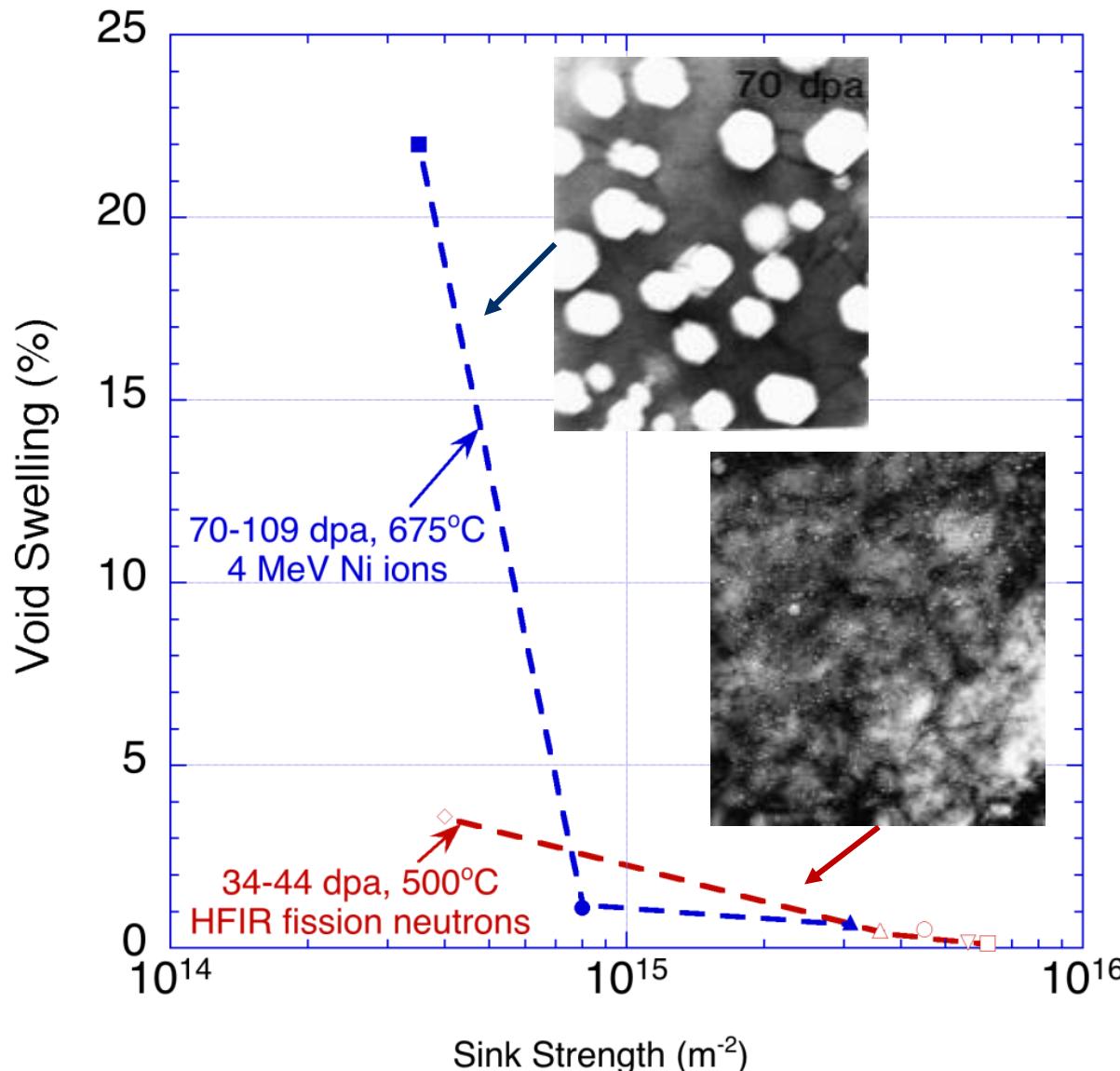


R. Yamada, S.J. Zinkle and G.P. Pells, J. Nucl. Mater. 209(1994) 191



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Design for Radiation Resistance III: High Sink Strength



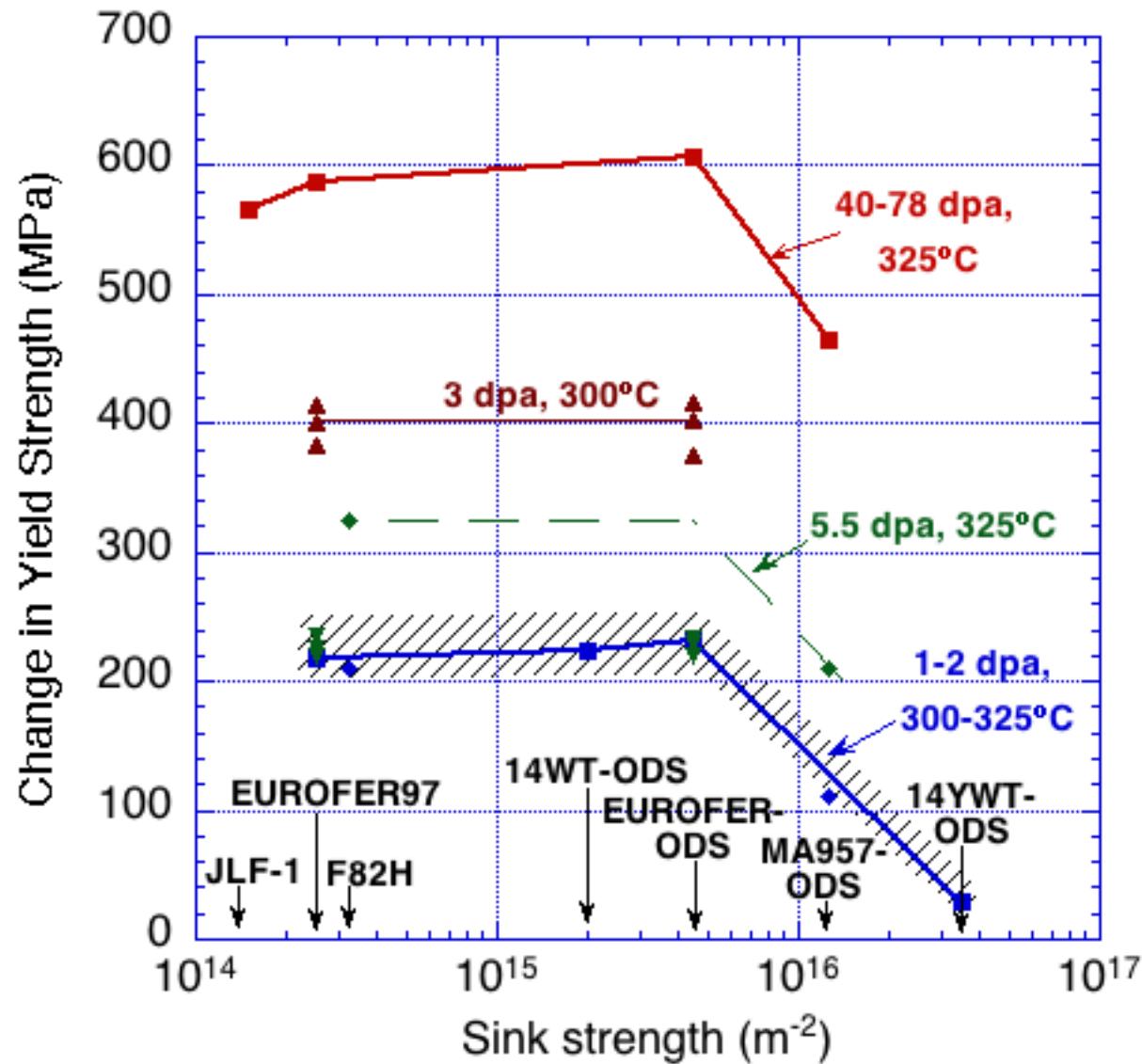
Dramatic reduction in void swelling occurs when average spacing between voids is $>10\times$ average spacing between defect sinks

$$N_v^{-1/3} \gg S_{\text{tot}}^{-1/2}$$

For void swelling resistance, sink strengths $>10^{15}/\text{m}^2$ are generally sufficient for fission reactors; fusion reactor irradiation may require even higher sink strengths ($>10^{16}/\text{m}^2?$) due to transmutant He production



Design for Radiation Resistance III: High Sink Strength



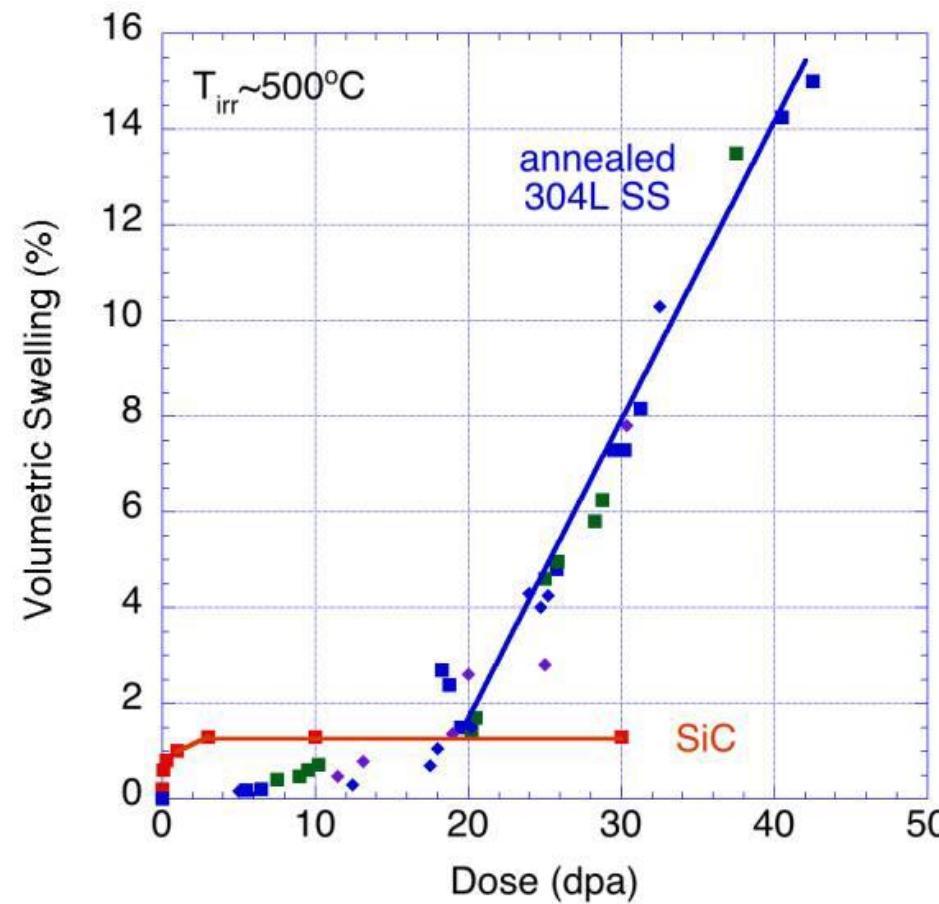
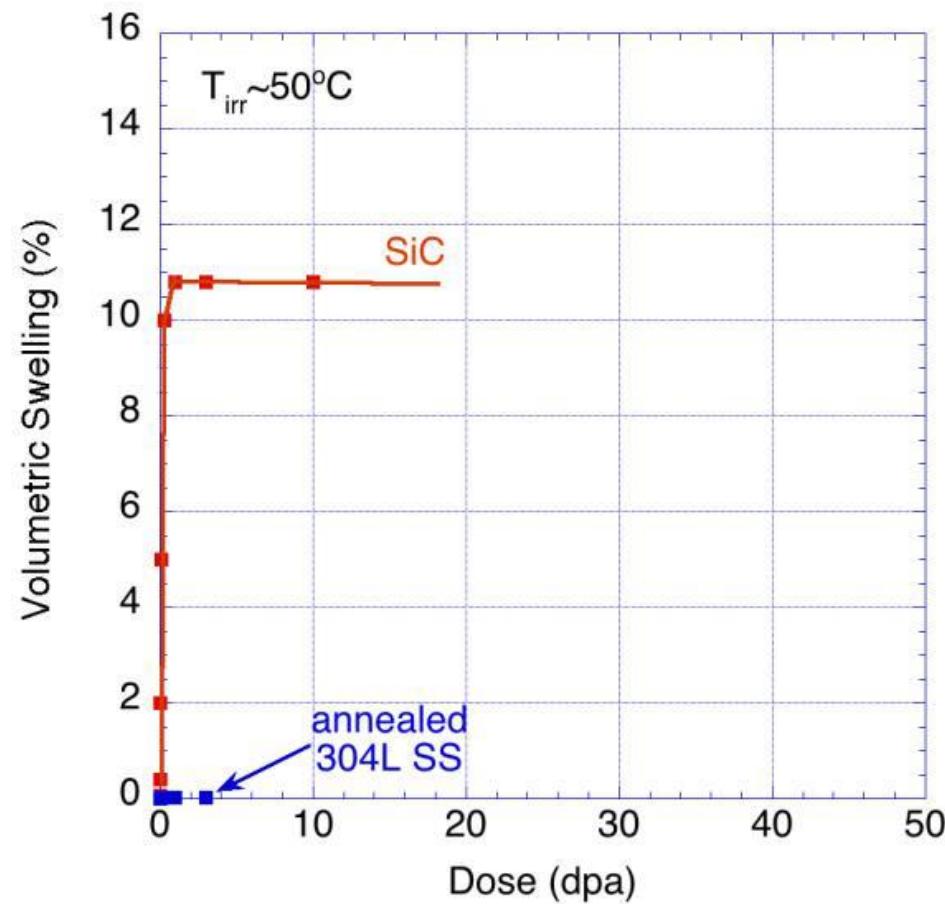
Dramatic reduction in radiation hardening occurs when average spacing between defect cluster nuclei (dislocation loops, etc.) is much greater than average spacing between defect sinks

$$N_{\text{loop}}^{-1/3} \gg S_{\text{tot}}^{-1/2}$$

-Or-

$$S_{\text{tot}} \gg S_{\text{rad defects}}$$

BUT, don't forget about irradiation regimes!

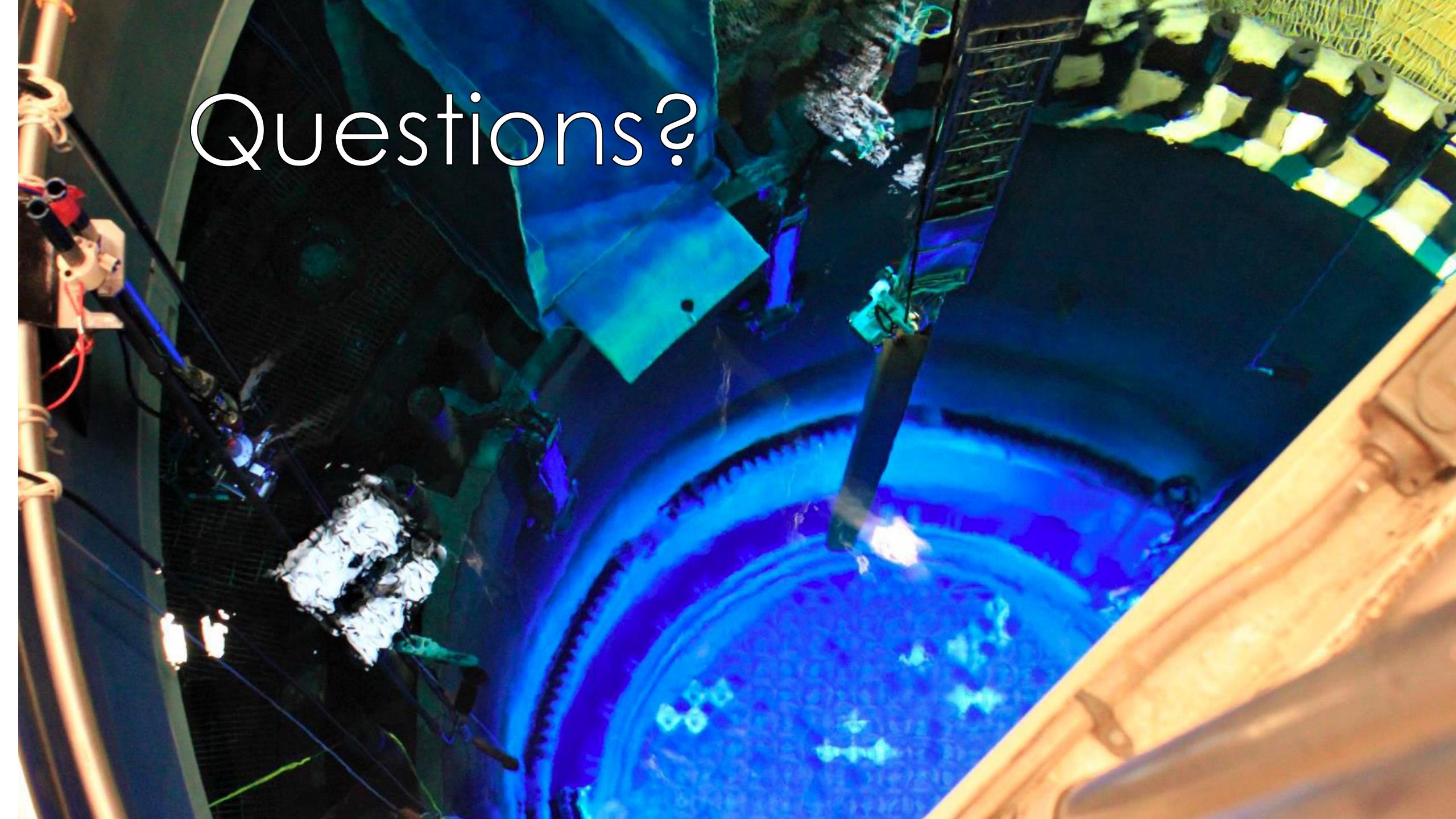


Swelling differences largely due to difference in melting temperatures (point defect mobility)

That's it!



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Questions?