

# Adsorption of Ammonia and Its Influence on Coadsorbed Carbon Monoxide on Monolayer and Multilayer Palladium Epitaxially Grown on Mo(110)

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The adsorption of ammonia on a monolayer of Pd on Mo(110), Pd<sub>1ML</sub>/Mo(110), and multilayer Pd(111)/Mo(110) surfaces has been studied using low-energy electron diffraction (LEED), temperature-programmed desorption (TPD), and high-resolution electron energy loss spectroscopy (HREELS). A diffuse (2 × 2) LEED pattern has been observed for ammonia on the Pd<sub>1ML</sub>/Mo(110) surface. TPD measurements indicate a shift of the desorption peak maximum of ammonia to a lower temperature for monolayer palladium compared to multilayer palladium. However, both surfaces display virtually identical HREELS spectra with the symmetric deformation (umbrella mode) being the dominating feature, indicating an upright adsorption geometry on both surfaces. The coadsorption of NH<sub>3</sub> and CO has been also studied with HREELS over a wide coverage range on both surfaces. A strong interaction between adsorbed NH<sub>3</sub> and CO is indicated by the red shift of the CO stretching vibrational mode (as large as 400 cm<sup>-1</sup>) and a blue shift of the NH<sub>3</sub> symmetric deformation (up to 100 cm<sup>-1</sup>). On Pd<sub>1ML</sub>/Mo(110), the coadsorption of NH<sub>3</sub> induces a CO adsorption site change from the atop site to the three-hollow site. Evidence has been also found for a significantly inclined NH<sub>3</sub> molecule in the coadsorbed layer with a relatively high CO coverage.

## 1. Introduction

The chemistry of bimetallic surfaces has received considerable attention in recent years due to the novel properties exhibited by these systems relative to their single-component counterparts.<sup>1–5</sup> Palladium monolayers supported on early transition metals are the most studied systems of this kind.<sup>1–17</sup> It has been found that a palladium monolayer has a reduced ability to adsorb CO, H<sub>2</sub>, and C<sub>2</sub>H<sub>4</sub> compared to bulk palladium.<sup>6–18</sup> There is some controversy in the literature regarding the origin of the modified properties of supported monolayer transition metals on dissimilar metals. Systematic studies have been carried out to address the correlation between the adsorption behavior of a supported metal overlayer toward CO and its electronic structure.<sup>1,4,5</sup> The change in the binding energy of CO on a number of supported metal monolayers has been correlated with the XPS core level shifts of the metal overlayer as well as work function changes.<sup>4,5</sup> A simple model has been advanced to explain the reduced binding energy of a Pd monolayer on early transition metal: the depletion of 4d electron density through charge polarization from Pd toward the metal substrate and the rehybridization of the Pd(4d,5s,5p) levels.<sup>1,5</sup>

Unlike CO, NH<sub>3</sub> is a weak donor ligand, which adsorbs on the surface through the nitrogen lone pair electrons in a single dative bond. It is instructive to compare these two different bonding mechanisms, especially multilayer palladium relative to monolayer palladium. The adsorption of NH<sub>3</sub> on transition-metal surfaces is also of considerable interest because of its relevance to ammonia synthesis. In recent years, the catalytic de-NO<sub>x</sub> process using NH<sub>3</sub> has also received considerable attention. The interaction of ammonia with NH<sub>3</sub>-synthesis-relevant surfaces, such as Fe and Ru, has been extensively characterized using a number of surface science techniques (refs

19–21 and references therein). These studies have been extended to other surfaces including Pt(111),<sup>22–24</sup> Ni(111),<sup>25–28</sup> Ni(110),<sup>29–30</sup> Ag(110),<sup>31</sup> Ag(311),<sup>32</sup> Al(111),<sup>33</sup> and NiO(100).<sup>34</sup> However, no studies have been carried out on palladium surfaces or any ultrathin metal overlayers. In this paper, we report on our recent studies of adsorption of NH<sub>3</sub> on palladium monolayers supported on the Mo(110) surface and a Pd(111) surface epitaxially grown on the Mo(110) surface.

The second focus of this paper is the coadsorption of NH<sub>3</sub> with CO on these surfaces. The coadsorption of NH<sub>3</sub> and CO has been studied previously on Ni(111),<sup>35–38</sup> Ni(110),<sup>36,37</sup> Ru(001),<sup>39,40</sup> Re(0001),<sup>41</sup> and Cu(110)<sup>42</sup> surfaces. Using metastable quenching spectroscopy (MQS), Tochihara et al.<sup>35</sup> have studied the coadsorption of NH<sub>3</sub> and CO on Ni(111). These authors found that NH<sub>3</sub> on top of a CO layer leads to an increase in the back-donation from the metal substrate into the 2π\* orbital of CO. The adsorption of NH<sub>3</sub> on top of CO chemisorbed on Ni(111) and Ni(110) has been studied by Dresser et al.<sup>34</sup> and Lanzillotto et al.,<sup>37</sup> using electron-stimulated desorption ion angular distribution (ESDIAD) and temperature-programmed desorption (TPD). A short-range CO–NH<sub>3</sub> interaction was found on Ni(111) with the C<sub>3v</sub> axis of NH<sub>3</sub> inclined away from the surface normal. Erley<sup>38</sup> has studied the coadsorption of NH<sub>3</sub> and CO on Ni(111) using infrared spectroscopy. Zhou et al.<sup>39</sup> and Sakaki et al.<sup>40</sup> studied NH<sub>3</sub> and CO coadsorption on Ru(001) using high-resolution electron energy loss spectroscopy (HREELS), low-energy electron diffraction (LEED), and TPD, and they showed that CO coadsorbed with NH<sub>3</sub> induces a red shift in the C–O stretching vibrational mode. Mijoule et al.<sup>43</sup> have carried out density functional calculations of the vibrational stretching mode of CO coadsorbed with NH<sub>3</sub> on Pd clusters. Their calculations showed that the C–O stretching frequency, the CO bond length, and the magnitude of the d–π\* back-donation are modified similarly whether NH<sub>3</sub> is added explicitly or electrons are simply added to the cluster. However, the

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metal–CO vibration, the metal–CO bond length, and the magnitude of the s–d donation are all altered in the opposite direction. Therefore, these authors concluded that the effects of the coadsorption of  $\text{NH}_3$  with CO cannot arise simply from electron transfer from  $\text{NH}_3$  to CO through the metal. Other more subtle effects must also be involved.

## 2. Experimental Section

The experiments have been carried out in a ultrahigh-vacuum (UHV) chamber which has been described in detail previously<sup>44</sup> and whose base pressure was  $<1 \times 10^{-10}$  Torr. Briefly, the UHV chamber is equipped with capabilities for HREELS (LK2000), Auger electron spectroscopy (AES), LEED, and TPD. HREEL spectra were acquired with a primary energy of 3 eV and a typical resolution of  $50\text{--}60\text{ cm}^{-1}$ . TPD measurements were made with the QMS in line-of-sight with the sample and with a linear heating rate of  $\sim 5\text{ K/s}$ . To avoid electron beam damage, the sample was biased at  $-100\text{ V}$  during the TPD experiments. The sample could be resistively heated to 1500 K or heated to 2200 K using an electron beam assembly. The sample temperature was measured using a W–5% Re/W–26% Re thermocouple spot-welded to the sample's edge. The Mo(110) crystal was cleaned by annealing in  $2 \times 10^{-8}$  Torr of  $\text{O}_2$  at 1200 K, with a subsequent flash to 2000 K. This procedure was repeated several times until no contaminations could be detected via AES.

The palladium source used for deposition was a 0.25 mm Pd wire (99.997%, Johnson Matthey Chemical Limited) wrapped around a tungsten filament. A line-of-sight mass spectrometer was used to monitor the Pd flux which typically corresponded to approximately one monolayer (ML) per minute. The Pd source has been extensively outgassed before use; the pressure during evaporation never exceeded  $3 \times 10^{-10}$  Torr. The Pd coverage was calibrated using TPD and AES. The growth and annealing behavior of Pd on Mo(110) was studied by Park et al.,<sup>45</sup> using AES, XPS, LEED, and the change in work function. At monolayer or less coverage, Pd forms a pseudomorphic monolayer with no indication of alloy formation. At higher coverages, Pd grows in a layer-by-layer or Frank–van der Merwe mode at room temperature. At elevated temperatures, Pd grows layer-by-layer in the first layer followed by three-dimensional (3D) clustering or a Stranski–Krastanow mechanism. Annealing the Mo(110) surface exposed to Pd vapor at room temperature to 600 K will also cause the Pd multilayer to agglomerate. Significant alloying of Pd with Mo via diffusion of Mo into Pd occurs; however, even with the alloying, the surface is still covered by a monolayer of Pd. If the Pd multilayer grown at room temperature exceeds 12 ML, annealing to temperatures as high as 900 K will not cause coalescence, apparently due to kinetic considerations. The thick Pd layer shows a sharp hexagonal LEED pattern, indicating epitaxial growth and the formation of a Pd(111) overlayer structure. Only monolayer palladium and multilayer ( $>15\text{ ML}$ ) Pd structures with sharp LEED patterns have been utilized in the present studies. For brevity, the Pd monolayer on Mo(110) and the epitaxially grown Pd multilayer on Mo(110) will be referred to as the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and  $\text{Pd}(111)/\text{Mo}(110)$  surfaces in the following sections.

The ammonia (Matheson, 99.99%) was used after several freeze–pump–thaw cycles in the gas manifold. Carbon monoxide (Matheson, 99.99%) was used as received without further purification. The gases were introduced via a directional gas doser to the surface for TPD measurements, while the exposures for the HREELS experiments were carried out via

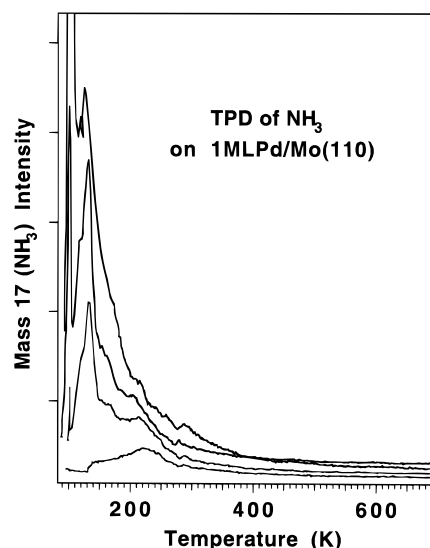


Figure 1. 1. TPD spectra of  $\text{NH}_3$  from the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  surface at various  $\text{NH}_3$  exposures.

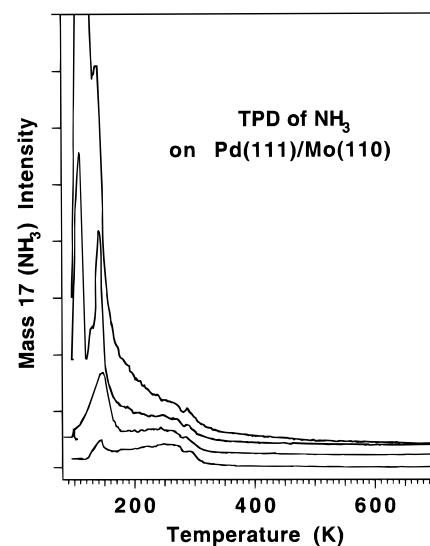
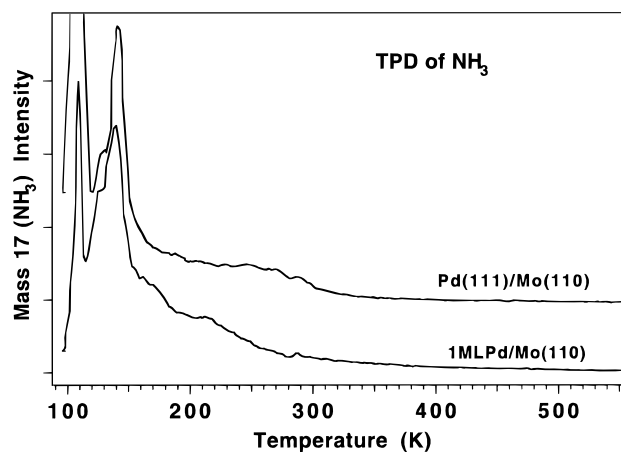


Figure 2. 2. TPD spectra of  $\text{NH}_3$  from the  $\text{Pd}(111)/\text{Mo}(110)$  surface at various  $\text{NH}_3$  exposures.

backfilling the UHV chamber. The exposures are given in langmuirs, equal to  $1 \times 10^{-6}$  Torr·s, without correction for the dose enhancement and ion gauge sensitivity. For low coverages, the exposures have been converted in certain cases to coverage using LEED and HREELS results and assuming a constant sticking coefficient.

## 3. Results

**3.1.  $\text{NH}_3$  Adsorption.** **3.1.1. LEED and TPD.** The adsorption of  $\text{NH}_3$  was first studied using TPD with masses 2, 17, 18, 28, and 32 routinely monitored. On both  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and  $\text{Pd}(111)/\text{Mo}(110)$  surfaces, molecular  $\text{NH}_3$  desorption is the only product found in the gas phase. Consistently, AES after TPD experiments showed a surface free of nitrogen, indicating that  $\text{NH}_3$  adsorbs molecularly and that the adsorption is completely reversible. Figures 1 and 2 show two series of TPD spectra acquired after increasing exposures of  $\text{NH}_3$  to  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and  $\text{Pd}(111)/\text{Mo}(110)$  surfaces, respectively. The TPD spectra of Figure 1 for the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  surface show desorption features between 140 and 350 K for low exposures of  $\text{NH}_3$ . Three broad and significantly overlapping peaks centered at 160,

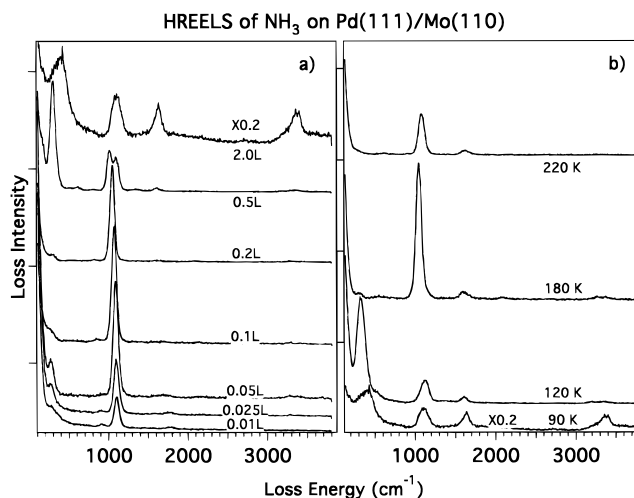


**Figure 3.** Comparison of TPD spectra on the Pd<sub>1ML</sub>/Mo(110) surface with the Pd(111)/Mo(110) surface at a NH<sub>3</sub> coverage of approximate 1.5 ML.

225, and 280 K are apparent. With increasing NH<sub>3</sub> exposures, the features between 150 and 350 K become more intense, and two new features at 100 and 130 K begin to develop. The feature at 100 K does not saturate but shifts gradually to higher desorption temperatures with increasing NH<sub>3</sub> dose. This feature is assigned to desorption of NH<sub>3</sub> from the multilayer. The feature at 130 K, with a desorption temperature between the multilayer and the first chemisorbed layer, is assigned to second-layer desorption. The Pd(111)/Mo(110) surface displays very similar TPD behavior as shown in Figure 2. Two desorption features between 90 and 150 K are observed, again attributed to desorption from the condensed and second layers, respectively. The chemisorbed layer, however, shows a slightly larger proportion of the high-temperature desorption state on the Pd(111)/Mo(110) surface compared to the Pd<sub>1ML</sub>/Mo(110) surface. This is more evident in Figure 3, where TPD spectra from the two surfaces are compared. From Figure 3 one can conclude that a significant fraction of the desorption feature near 270 K on the Pd(111)/Mo(110) surface is shifted to a lower desorption temperature on the Pd<sub>1ML</sub>/Mo(110) surface. However, the high-temperature state near 270 K is still populated on the Pd<sub>1ML</sub>/Mo(110) surface, but with reduced intensity.

The adsorption of NH<sub>3</sub> on these two surfaces has been further studied with LEED. On the Pd(111)/Mo(110) surface, no additional LEED spots were observed upon NH<sub>3</sub> exposures and subsequent annealing. In contrast, the Pd<sub>1ML</sub>/Mo(110) surface exhibited a (2 × 2) LEED pattern after annealing to 180 K after dosing with approximately 3 ML of NH<sub>3</sub>. This temperature is sufficient to remove both multilayer and second-layer ammonia, leaving a surface covered with only a chemisorbed layer of ammonia. The LEED pattern is very sensitive to the electron beam and can be observed only for a relatively short time period, indicating a high cross section of electron-induced desorption for NH<sub>3</sub> on the Pd<sub>1ML</sub>/Mo(110) surface. This is consistent with the relative weak bonding of NH<sub>3</sub> on this surface. The (2 × 2) LEED pattern observed here has been previously observed by Benndorf and Madey<sup>46</sup> for NH<sub>3</sub> on the Ru(001) surface but could not be confirmed by Sasaki et al.<sup>40</sup>

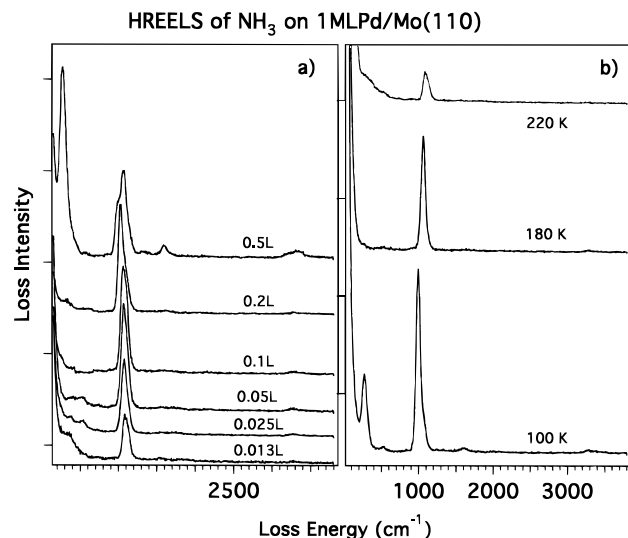
**3.1.2. HREELS.** Figure 4a shows a series of HREEL spectra acquired subsequent to increasing NH<sub>3</sub> exposures on the Pd(111)/Mo(110) surface. At an exposure of 0.01 langmuir, two features are seen at 290 and 1100 cm<sup>-1</sup>, which can be assigned to the Pd–N stretching and symmetric deformation (umbrella) modes of adsorbed NH<sub>3</sub>, respectively. No features related to the N–H stretching were observed, presumably due to the low cross section for these modes. Upon increasing the NH<sub>3</sub>



**Figure 4.** HREEL spectra of NH<sub>3</sub> on the Pd(111)/Mo(110) surface: (a) after increasing exposures, (b) after dosing ca. 4 ML of NH<sub>3</sub> at 90 K and subsequently annealing to the indicated temperature.

exposure from 0.01 langmuir, the features at 290 and 1100 cm<sup>-1</sup> gain intensity until 0.05 langmuir is reached; the 290 cm<sup>-1</sup> feature then loses intensity with a further increase in exposure to 0.2 langmuir. Concomitantly, a new feature at 300 cm<sup>-1</sup> begins to appear at 0.2 langmuir. The intensity of the peak at 1100 cm<sup>-1</sup> is almost constant between 0.05 and 0.1 langmuir and then begins to attenuate with exposures up to 0.2 langmuir. The symmetric deformation mode gradually shifts from 1100 to 1050 cm<sup>-1</sup> with increasing NH<sub>3</sub> exposures from 0.01 to 0.2 langmuir. At an exposure of 0.025 langmuir, features between 3200 and 3500 cm<sup>-1</sup>, assigned to N–H stretching modes, begin to develop. With increasing NH<sub>3</sub> exposure from 0.2 to 0.5 langmuir, the feature at 1050 cm<sup>-1</sup> shifts further to 1005 cm<sup>-1</sup> and dramatically loses intensity. Concomitantly, two new features appear at 1100 and 1610 cm<sup>-1</sup>, while the feature at 300 cm<sup>-1</sup> dominates at this coverage. These features collectively can be assigned to bilayer formation. The peaks at 300 and 1610 cm<sup>-1</sup> correspond to the NH<sub>3</sub> twisting mode and the asymmetric deformation mode, respectively. The NH<sub>3</sub> twisting mode is very characteristic of second-layer ammonia on various surfaces and has been used as a fingerprint for the onset of second-layer ammonia adsorption.<sup>20</sup> This mode is not visible in the first chemisorbed layer due to free rotation about the N–metal bond.<sup>46</sup> In the second ammonia layer, the formation of hydrogen bonds between the ammonias hinders this rotation considerably and shifts its frequency to an accessible frequency in the HREELS measurements. The top spectrum of Figure 4a is that acquired after dosing 2 langmuirs of NH<sub>3</sub> (approximately 5–6 monolayers). In this spectrum, features at 420, 1100, and 1625 cm<sup>-1</sup> and a doublet at 3360 and 3400 cm<sup>-1</sup> are apparent. These features correlate with the formation of the NH<sub>3</sub> multilayer phase and can be assigned to the frustrated libration (twisting mode), the symmetric and asymmetric deformation, and the symmetric and asymmetric stretching vibrational modes, respectively.

In Figure 4b, an annealing series after dosing approximately four NH<sub>3</sub> layers on the Pd(111)/Mo(110) surface at 90 K is shown. After heating to 120 K to remove the multilayer, the spectrum characteristic of the bilayer is seen, with dominating features at 320 cm<sup>-1</sup> due to the NH<sub>3</sub> twisting mode and the symmetric and asymmetric deformation mode at 1120 and 1600 cm<sup>-1</sup>, respectively. Further annealing to 180 K desorbs the second layer with the subsequent spectrum being dominated by the symmetric deformation at 1050 cm<sup>-1</sup>. Also seen are the



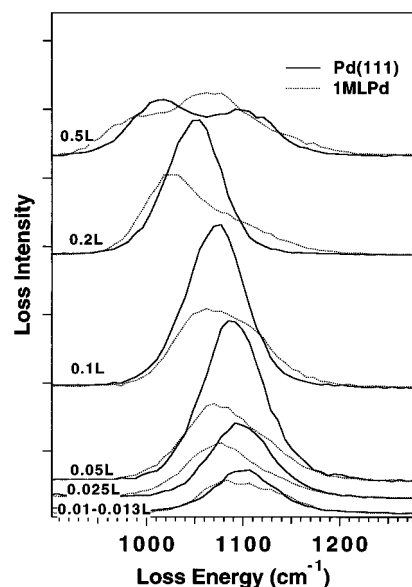
**Figure 5.** 5. HREEL spectra of  $\text{NH}_3$  on the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  surface: (a) after increasing exposures, (b) after dosing about 4 ML of  $\text{NH}_3$  at 90 K and subsequently annealing to the indicated temperature.

$\text{Pd-N}$  vibrational mode at  $300\text{ cm}^{-1}$  and the asymmetric and symmetric  $\text{N-H}$  stretching modes at  $3260$  and  $3365\text{ cm}^{-1}$ , respectively. A small feature at  $1600\text{ cm}^{-1}$  is also apparent, which is most likely due to a trace amount of coadsorbed  $\text{CO}$ . This feature is less likely due to the asymmetric deformation of  $\text{NH}_3$ , since the asymmetric deformation mode is not observed at a similar coverage in the coverage series of Figure 4a, whereas  $\text{CO}$  coadsorbed with  $\text{NH}_3$  shows a  $\text{C-O}$  vibrational mode at this frequency (see below). Upon heating the sample to even higher temperatures, the feature at  $1050\text{ cm}^{-1}$  loses intensity and shifts to a higher wavenumber due to desorption of  $\text{NH}_3$  from the chemisorbed first layer.

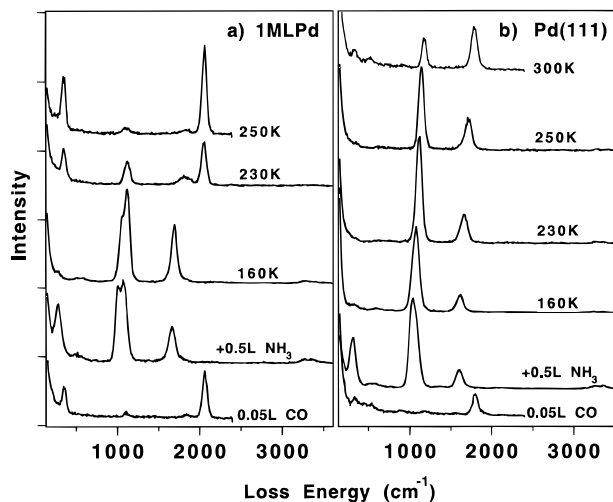
A palladium monolayer supported on  $\text{Mo}(110)$ ,  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$ , exhibits HREEL spectra analogous to those of the  $\text{Pd}(111)/\text{Mo}(110)$ . This can be seen in Figure 5a,b. In Figure 5a, the HREEL spectra acquired after increasing  $\text{NH}_3$  exposures are shown, while Figure 5b shows an annealing series. As for  $\text{Pd}(111)/\text{Mo}(110)$ , the symmetric  $\text{NH}_3$  deformation, which gradually gains intensity and shifts to lower wavenumber with increasing  $\text{NH}_3$  coverage, is the dominating feature up to a  $\text{NH}_3$  exposure of 0.2 langmuir. The intensity of this feature decreases with further increasing exposure to 0.5 langmuir. The features correlated to the bilayer  $\text{NH}_3$  dominate the spectrum at 0.5 langmuir. However, careful inspection of the symmetric deformation shows some differences on the two surfaces that are more evident in Figure 6. On the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  surface, the features are generally broad and shift more rapidly to lower wavenumbers with increasing  $\text{NH}_3$  exposures. The intensity change with increasing  $\text{NH}_3$  is more profound for the  $\text{Pd}(111)/\text{Mo}(110)$  surface than the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  surface. Also, an asymmetric peak shape is apparent for the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  surface.

**3.2. Coadsorption of  $\text{NH}_3$  with  $\text{CO}$ .** The influence of  $\text{NH}_3$  on the vibrational structure of coadsorbed  $\text{CO}$  on the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and  $\text{Pd}(111)/\text{Mo}(110)$  surface has been studied using HREELS over a wide coverage range. In the following sections these results are described and the two surfaces compared at various coverages.

**3.2.1. 0.05 langmuir of  $\text{CO}$  + 0.5 langmuir of  $\text{NH}_3$ .** Figure 7 shows HREEL spectra for the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  (left) and  $\text{Pd}(111)/\text{Mo}(110)$  (right) systems at a relatively small  $\text{CO}$  and a relatively large  $\text{NH}_3$  coverage. From bottom to top, the spectra



**Figure 6.** 6. Comparison of the  $\text{NH}_3$  symmetric deformational mode on the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  surface with the  $\text{Pd}(111)/\text{Mo}(110)$  surface at various  $\text{NH}_3$  coverages.



**Figure 7.** 7. HREEL spectra after dosing 0.5 langmuir of  $\text{NH}_3$  with 0.05 langmuir of  $\text{CO}$  precovered on (a)  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and (b)  $\text{Pd}(111)/\text{Mo}(110)$  surfaces and subsequently annealing to the indicated temperature.

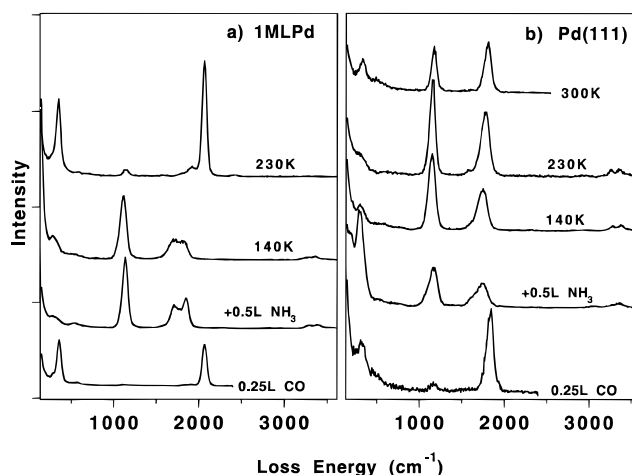
acquired after dosing 0.05 langmuir of  $\text{CO}$ , then 0.5 langmuir of  $\text{NH}_3$ , and finally an anneal to the indicated temperature are shown. The spectra taken for a 0.05 langmuir  $\text{CO}$  exposure show features at  $350$  and  $2050\text{ cm}^{-1}$  on the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  surface and features at  $320$  and  $1800\text{ cm}^{-1}$  on the  $\text{Pd}(111)/\text{Mo}(110)$  surface. These can be assigned to  $\text{CO}$  adsorption on the atop and 3-fold hollow sites, respectively. Dosing 0.5 langmuir of  $\text{NH}_3$  on top of these surfaces induces a dramatic red shift of the  $\text{C-O}$  stretching vibration by  $400\text{ cm}^{-1}$  for  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and  $200\text{ cm}^{-1}$  for  $\text{Pd}(111)/\text{Mo}(110)$ . Concomitantly, the features due to the  $\text{Pd-CO}$  stretching mode are replaced by the  $\text{NH}_3$  twisting mode arising from  $\text{NH}_3$  in the second layer. The  $\text{NH}_3$  symmetric deformational mode exhibits a doublet at  $1010$  and  $1080\text{ cm}^{-1}$  for  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and a single, broad peak at  $1030\text{ cm}^{-1}$  for  $\text{Pd}(111)/\text{Mo}(110)$ . Also, the asymmetric and symmetric  $\text{N-H}$  stretching modes are clearly seen for both surfaces. Since the  $\text{NH}_3$  desorbs at a lower temperature than  $\text{CO}$ , the coadsorption of  $\text{NH}_3$  at relatively low coverages with the same  $\text{CO}$  surface coverage has been studied by annealing the surface to the indicated temperature in Figure 7. The surface

was first annealed to 160 K to remove the  $\text{NH}_3$  second layer, and as anticipated the peak near  $300\text{ cm}^{-1}$  essentially disappears. Only a small feature, likely relating to the Pd–N stretching mode, remains near  $300\text{ cm}^{-1}$ . It should be noted that the Pd–CO stretching mode is not observed for either surface, although this mode has a very high intensity on the  $\text{NH}_3$ -free  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface. A small shift ( $10\text{--}15\text{ cm}^{-1}$ ) in both the  $\text{NH}_3$  symmetric deformational mode and the CO stretching vibrational mode to higher wavenumber is also apparent in this temperature range. On the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface, the doublet near  $1050\text{ cm}^{-1}$  is replaced by a feature at  $1010\text{ cm}^{-1}$  with a shoulder on the low wavenumber side. The HREEL spectra of  $\text{NH}_3$  alone on the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  and  $\text{Pd}(111)/\text{Mo}(110)$  surfaces show a  $\text{NH}_3$  symmetrical, deformational mode for the first chemisorbed layer below  $1005\text{ cm}^{-1}$  and, for the second layer, a feature between  $1065$  and  $1100\text{ cm}^{-1}$  at  $0.5$  langmuir of  $\text{NH}_3$ . Comparing the results of coadsorbed  $\text{NH}_3$  and CO with  $\text{NH}_3$  alone, it is apparent that the presence of CO on the surface shifts the  $\text{NH}_3$  symmetrical deformational mode for the first chemisorbed layer of  $\text{NH}_3$  to a higher frequency. For the  $\text{Pd}(111)/\text{Mo}(110)$  surface, this shift causes the feature for the first chemisorbed layer to collapse into the feature for the second adsorbed layer, resulting in a single  $\text{NH}_3$  symmetric deformation peak.

To further reduce the  $\text{NH}_3$  coverage, the sample was heated from 160 to 300 K, and HREEL spectra were acquired after anneals to 230, 250, and 300 K. In this temperature range, the  $\text{NH}_3$  symmetric deformation and the C–O stretching vibration gradually shift to higher wavenumbers on the  $\text{Pd}(111)/\text{Mo}(110)$  surface. For the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface the C–O stretching mode moves to  $2050\text{ cm}^{-1}$  with a small feature at  $1800\text{ cm}^{-1}$  becoming apparent at 230 K. Also seen is the reappearance of the Pd–CO stretching mode at  $350\text{ cm}^{-1}$  for the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface. The disappearance and reappearance of this peak correlate with the absence and presence of CO adsorbed on the atop site. The coadsorption of  $\text{NH}_3$  shifts the CO adsorption from the atop site to the three-hollow site. The Pd–CO vibrational mode of CO adsorbed on the three-hollow site apparently has a smaller intensity and lower vibrational frequency than does CO adsorbed on the atop site and is obscured by the  $\text{NH}_3$  twisting mode.

On the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface, the features related to  $\text{NH}_3$  are greatly reduced upon heating to 250 K after which the C–O stretching mode moves to its position prior to the dosing of  $\text{NH}_3$ . Also, the peak desorption temperature of  $\text{NH}_3$  is much higher for the  $\text{Pd}(111)$  surface. Even at 300 K, significant amounts of  $\text{NH}_3$  remain on the surface, as indicated by the feature at  $1170\text{ cm}^{-1}$ . These results show a higher binding energy for  $\text{NH}_3$  on the  $\text{Pd}(111)/\text{Mo}(110)$  surface than for the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface, even in the presence of CO. Furthermore, by comparing the anneal series with coadsorbed  $\text{NH}_3$  to that acquired on the CO-free surface, it can be concluded that coadsorption of  $\text{NH}_3$  with CO increases the  $\text{NH}_3$  binding energy. This conclusion is also valid for other CO coverages, as discussed below.

**3.2.2. 0.25 langmuir of CO + 0.5 langmuir of  $\text{NH}_3$ .** Figure 8 shows two series of HREEL spectra acquired after dosing 0.5 langmuir of  $\text{NH}_3$  to the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  and  $\text{Pd}(111)/\text{Mo}(110)$  surfaces, precovered with 0.25 langmuir of CO, and a subsequent anneal of each to the indicated temperature. For reference, the spectra of the surfaces with only CO are shown at the bottom. The peaks at  $360$  and  $2060\text{ cm}^{-1}$  on the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  and at  $320$  and  $1830\text{ cm}^{-1}$  on the  $\text{Pd}(111)/\text{Mo}(110)$  surface can be assigned to the Pd–CO and C–O stretching modes, respectively. The adsorption sites are atop for the  $\text{Pd}_{\text{IML}}/$

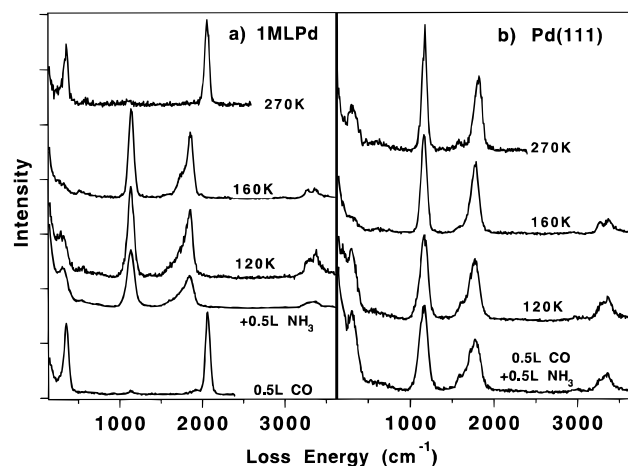


**Figure 8.** HREEL spectra after dosing 0.5 langmuir of  $\text{NH}_3$  with 0.25 langmuir of CO-precovered (a)  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  and (b)  $\text{Pd}(111)/\text{Mo}(110)$  surfaces and subsequently annealing to the indicated temperature.

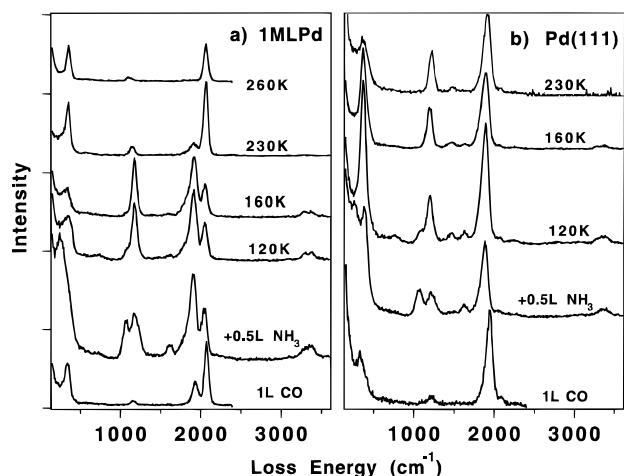
$\text{Mo}(110)$  surface and likely three-hollow for the  $\text{Pd}(111)/\text{Mo}(110)$  surface. Dosing 0.5 langmuir of  $\text{NH}_3$  on top of these surfaces shifts the C–O stretching vibration to lower wavenumber. On the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface, two features corresponding to C–O stretching modes appear at  $1705$  and  $1810\text{ cm}^{-1}$ , while on the  $\text{Pd}(111)/\text{Mo}(110)$  surface a feature at  $1750\text{ cm}^{-1}$  with a shoulder at  $1600\text{ cm}^{-1}$  is seen. As for lower CO coverages, the features corresponding to the Pd–CO stretching modes are not observed. Instead, the  $\text{NH}_3$  twisting modes are observed in this frequency range. It is noteworthy that this twisting mode is much weaker for the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface compared to the  $\text{Pd}(111)/\text{Mo}(110)$  surface or to the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface with a lower CO coverage. This is probably due to the lower capacity of the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface for the adsorption of CO and  $\text{NH}_3$ . At an exposure of 0.25 langmuir of CO, the amount of  $\text{NH}_3$  adsorbed in the first chemisorbed layer on the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  is smaller than on the  $\text{Pd}(111)/\text{Mo}(110)$  surface. The bilayer structure for  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  is therefore limited to a relatively small part of the surface, suggesting that the twisting mode at  $300\text{ cm}^{-1}$  is characteristic of  $\text{NH}_3$  on top of  $\text{NH}_3$ . This is consistent with the observation that hydrogen bonding shifts this vibrational feature to an accessible value for detection by HREELS.

To examine the effect of  $\text{NH}_3$  coverage on CO adsorption at 0.5 langmuir, both surfaces were gradually annealed to 300 K. Following the desorption of the second  $\text{NH}_3$  layer by annealing to 140 K, the features corresponding to the  $\text{NH}_3$  twisting mode lose intensity. The  $\text{NH}_3$  symmetric deformational mode and the C–O stretching mode also show changes in the line shape without a significant shifting of the primary features. No significant shifts were observed for the  $\text{NH}_3$  symmetric deformational mode even at the higher annealing temperatures. However, the C–O stretching mode shifts from  $1750$  to  $1800\text{ cm}^{-1}$  for the  $\text{Pd}(111)/\text{Mo}(110)$  surface and from  $1705$  or  $1810\text{ cm}^{-1}$  to  $2060\text{ cm}^{-1}$  on the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface. A small feature at  $1910\text{ cm}^{-1}$  is also seen on the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface after annealing the surface to 230 K, which is probably due to CO interacting with small amounts of  $\text{NH}_3$  remaining on the surface. After desorbing essentially all of the  $\text{NH}_3$ , the Pd–CO stretching mode regains its original intensity.

**3.2.3. 0.5 langmuir of CO + 0.5 langmuir of  $\text{NH}_3$ .** The most striking finding in this coverage range, as shown in Figure 9, is the C–O stretching mode at  $1850\text{ cm}^{-1}$  for  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  and at  $1780\text{ cm}^{-1}$  for  $\text{Pd}(111)/\text{Mo}(110)$ . Each of these features



**Figure 9.** HREEL spectra after dosing 0.5 langmuir of  $\text{NH}_3$  onto 0.5 langmuir of CO-precovered (a)  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and (b)  $\text{Pd}(111)/\text{Mo}(110)$  surfaces and subsequently annealing to the indicated temperature.



**Figure 10.** HREEL spectra after dosing 0.5 langmuir of  $\text{NH}_3$  onto 1.0 langmuir of CO-precovered (a)  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and (b)  $\text{Pd}(111)/\text{Mo}(110)$  surfaces and subsequently annealing to the indicated temperature.

exhibits a foot extending with gradually decreasing intensity to  $1500\text{ cm}^{-1}$ . These results suggest a long-range interaction between coadsorbed CO and  $\text{NH}_3$  on both surfaces. Also noteworthy is that the main C–O stretching feature does not shift with annealing temperature to 160 K and that the  $\text{NH}_3$  symmetric deformational mode does not shift during the entire annealing cycle of these experiments.

As for lower CO coverages, the Pd–CO vibrational mode is quenched in the presence of  $\text{NH}_3$ , with the  $\text{NH}_3$  twisting mode becoming apparent. The N–H stretching mode is also observed with a relatively high intensity.

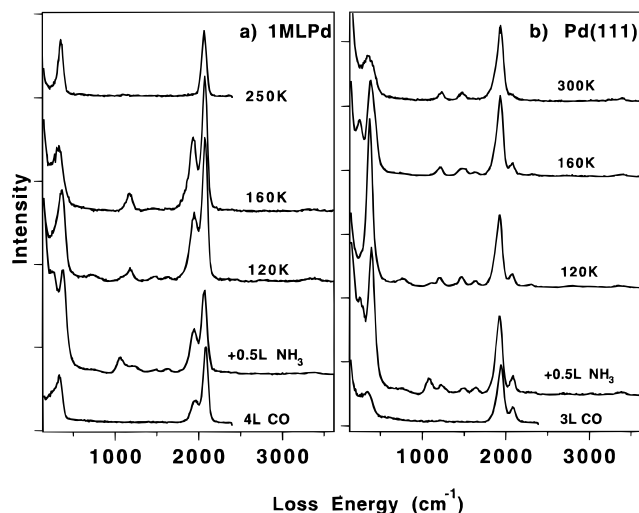
**3.2.4. High (1.0 langmuir) CO Coverage and 0.5 langmuir of  $\text{NH}_3$ .** The influence of  $\text{NH}_3$  on CO coadsorption at high CO coverage is shown in the HREEL spectra of Figure 10. The spectra of CO on the  $\text{NH}_3$ -free surfaces correspond to CO adsorption on the bridging site of  $\text{Pd}(111)/\text{Mo}(110)$  and both bridging and atop sites on  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$ . Dosing 0.5 langmuir of  $\text{NH}_3$  on top of the CO-precovered surface causes again rather dramatic changes in the C–O and Pd–CO vibrational regions.

For the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  surface, the intensity of the C–O vibrational mode shifts from a feature corresponding to the atop site to a feature related to the bridging site, with a small shift of both features to lower wavenumber. The peak at  $1905\text{ cm}^{-1}$

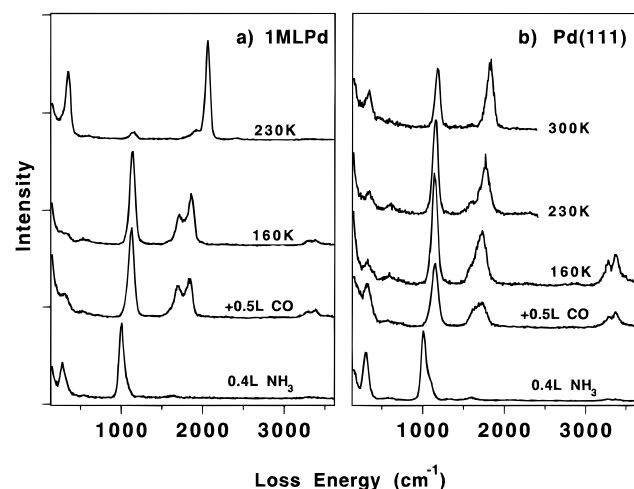
shows as before a foot with decreasing intensity extending to lower wavenumber. The feature at  $330\text{ cm}^{-1}$  corresponding to the Pd–CO stretching mode disappears into the shoulder of the feature corresponding to the  $\text{NH}_3$  twisting mode. Two well-separated features related to the  $\text{NH}_3$  symmetric deformational modes are observed at  $1080$  and  $1180\text{ cm}^{-1}$ . In contrast to the spectra acquired at lower CO coverages, a feature due to the  $\text{NH}_3$  asymmetric deformational mode is apparent at  $1610\text{ cm}^{-1}$ . Upon heating to 120 K, the feature at  $250\text{ cm}^{-1}$  disappears and the Pd–CO vibration at  $350\text{ cm}^{-1}$  regains some of the intensity; the latter's intensity is completely restored by heating to 230 K. The feature at  $1080\text{ cm}^{-1}$  loses significant intensity upon heating to 120 K and then disappears completely at 160 K with desorption of the  $\text{NH}_3$  second layer. This indicates that the large blue shift of the  $\text{NH}_3$  symmetric deformational mode is due to a surface-mediated interaction rather than a direct through-space interaction between CO and  $\text{NH}_3$ . Otherwise, one would expect to see a large shift as well for  $\text{NH}_3$  adsorbed in the second layer. It should be noted that the  $\text{NH}_3$  symmetric deformational mode for the first chemisorbed layer is at a higher wavenumber than that for the second layer. This result contrasts with that for the surfaces covered with only  $\text{NH}_3$ , where the first chemisorbed layer displays a lower vibrational frequency for the  $\text{NH}_3$  symmetric deformational mode (between  $970$  and  $1010\text{ cm}^{-1}$ ) than the second adsorbed  $\text{NH}_3$  layer (between  $1065$  and  $1100\text{ cm}^{-1}$ ). The gradual shift of the  $\text{NH}_3$  symmetric deformational mode to higher wavenumber with increasing CO coverages is the result of two processes. First, the presence of CO on the surface reduces the  $\text{NH}_3$  density in the first layer, therefore reducing the  $\text{NH}_3$ – $\text{NH}_3$  repulsive interaction. Second, the attractive interaction between  $\text{NH}_3$  and CO shifts the  $\text{NH}_3$  symmetric deformational mode to a higher frequency. Since the  $\text{NH}_3$  symmetric deformation shows a vibrational frequency of approximately  $1100\text{ cm}^{-1}$  at the zero-coverage limit with no CO coadsorbate, well below the vibrational frequency ( $1180\text{ cm}^{-1}$ ) with coadsorbed CO, the shift induced by the  $\text{NH}_3$ –CO attractive interaction contributes most significantly to the observed shift in the frequency of the  $\text{NH}_3$  symmetric deformation.

Coadsorption of CO with  $\text{NH}_3$  slightly shifts the C–O stretching vibration to lower frequency from  $1940$  to  $1890\text{ cm}^{-1}$ . Two  $\text{NH}_3$  symmetric deformation features are observed at  $1160$  and  $1210\text{ cm}^{-1}$ , with the lower frequency feature disappearing after annealing to 160 K. Both the Pd–CO and the  $\text{NH}_3$  twisting vibrational modes are evident. A new feature at  $1480\text{ cm}^{-1}$ , which was not observed at lower CO coverages, is apparent and becomes more evident upon annealing the surface to 120 K. This feature is still evident even after heating the sample to 230 K. This feature obviously arises from the relatively high coverage of CO. To confirm this, HREEL spectra were acquired after dosing 0.5 langmuir of  $\text{NH}_3$  onto a 3 langmuirs CO-precovered  $\text{Pd}(111)/\text{Mo}(110)$  surface. The resulting HREEL spectra are shown in Figure 11. As anticipated, the feature at  $1490\text{ cm}^{-1}$  is now even more intense. The low frequency of this feature precludes a CO-related origin. This peak is assigned accordingly to the asymmetric deformation of  $\text{NH}_3$ . The high intensity of this peak suggests that  $\text{NH}_3$  is no longer present in an upright configuration.

**3.2.5.  $\text{NH}_3$  + CO.** The effect of CO on coadsorbed  $\text{NH}_3$  can be more clearly seen if one doses CO onto an  $\text{NH}_3$ -precovered surface. HREEL spectra acquired after dosing 0.5 langmuir of CO on a 0.4 langmuir  $\text{NH}_3$ -precovered surface are shown in Figure 12. Spectra for  $\text{NH}_3$  alone are shown at the bottom of (a) and (b). Each surface was covered slightly in excess of 1 ML of  $\text{NH}_3$ , as indicated by the feature at  $300\text{ cm}^{-1}$  and the



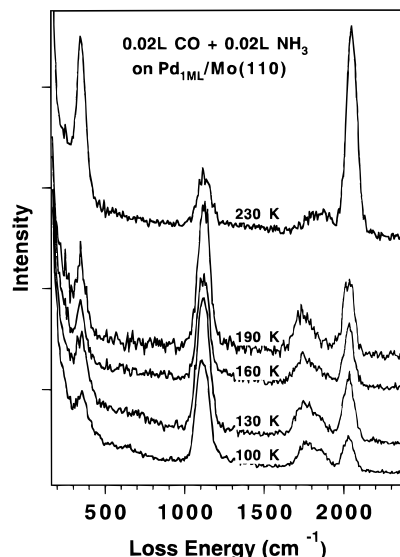
**Figure 11.** HREEL spectra after dosing 0.5 langmuir of  $\text{NH}_3$  onto 4.0 langmuir of CO-precovered (a)  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and (b)  $\text{Pd}(111)/\text{Mo}(110)$  surfaces and subsequently annealing to the indicated temperature.



**Figure 12.** HREEL spectra after dosing 0.5 langmuir of CO onto 0.4 langmuir of  $\text{NH}_3$ -precovered (a)  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and (b)  $\text{Pd}(111)/\text{Mo}(110)$  surfaces and subsequently annealing to the indicated temperature.

small shoulder at  $1100\text{ cm}^{-1}$ . Coadsorption with CO shifts the feature at  $1000\text{ cm}^{-1}$  to  $1140\text{ cm}^{-1}$ . The feature at  $300\text{ cm}^{-1}$  loses some intensity, likely to displacement of  $\text{NH}_3$  in the chemisorbed layer by CO. The intensity of the N–H stretching vibrations, however, is enhanced by the presence of CO. Two peaks corresponding to C–O stretching modes are seen at  $1700$  and  $1850\text{ cm}^{-1}$  for  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and at  $1610$  and  $1720\text{ cm}^{-1}$  for  $\text{Pd}(111)/\text{Mo}(110)$ . Upon annealing, the  $\text{NH}_3$  symmetric deformational mode loses intensity without shifting significantly, while the C–O stretching modes gradually shift to higher wavenumber.

**3.2.6. 0.02 langmuir of CO + 0.02 langmuir of  $\text{NH}_3$ .** The interaction of CO and  $\text{NH}_3$  at the zero-coverage limit has also been studied after dosing 0.02 langmuir of CO and 0.02 langmuir of  $\text{NH}_3$  on  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$ , corresponding to less than 10% of the saturation coverage for each species. The HREELS results are shown in Figure 13. From bottom to top, the surface has been annealed to higher temperatures as indicated. At 90 K, features at  $345$ ,  $1105$ ,  $1750$ ,  $1850$ , and  $2010\text{ cm}^{-1}$  are apparent and can be attributed to the Pd–CO vibration, the  $\text{NH}_3$  deformation, and the CO stretching mode for the three-hollow,



**Figure 13.** HREEL spectra after dosing 0.02 langmuir of CO onto 0.02 langmuir of  $\text{NH}_3$  precovered on  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and subsequently annealing to the indicated temperature.

the 2-fold bridging, and the atop site, respectively. Annealing to 230 K leads to a dramatic decrease in the  $\text{NH}_3$  deformation mode and an intensity transfer from the feature at  $1750\text{ cm}^{-1}$  to that at  $2010\text{ cm}^{-1}$ , indicating desorption of  $\text{NH}_3$  and a change of CO from the 3-fold hollow to the atop site. This evidence of significant interaction at such small coverages indicates the formation of mixed chemisorbed layers of CO and  $\text{NH}_3$ . This is consistent with an attractive interaction between CO and  $\text{NH}_3$  and repulsive CO–CO and  $\text{NH}_3$ – $\text{NH}_3$  interactions.

## 4. Discussion

**4.1.  $\text{NH}_3$  Adsorption.** The adsorption of  $\text{NH}_3$  has been previously studied on various surfaces using TPD.<sup>19–34</sup> Consistent with these previous results, the TPD spectra of  $\text{NH}_3$  in the present study show a broad desorption peak for the chemisorbed state on both the  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and  $\text{Pd}(111)/\text{Mo}(110)$  surfaces. This is due to the increasing repulsive interaction between the adsorbed  $\text{NH}_3$  with increasing coverage.  $\text{NH}_3$  adsorbs via a dative bond through the electron lone pair on the N atom, leading to a relatively large dipole between  $\text{NH}_3$  and the surface. These dipoles interact repulsively with each other, therefore destabilizing the  $\text{NH}_3$ . This general picture is consistent with work function measurements of  $\text{NH}_3$  adsorbed on  $\text{Ru}(001)$  by Benndorf and Madey.<sup>46</sup> These authors found that the work function initially decreases linearly with increasing  $\text{NH}_3$  coverage due to electron donation from  $\text{NH}_3$  to the surface but then decreases less rapidly up to 0.15 ML due to depolarization. The other possible explanation for the broad desorption feature observed for chemisorbed  $\text{NH}_3$  is the presence of multiple adsorption sites.<sup>23,32</sup> However, the single narrow feature observed for the  $\text{NH}_3$  symmetric deformational mode for the chemisorbed  $\text{NH}_3$  layer on  $\text{Pd}(111)/\text{Mo}(110)$  indicates that this is unlikely. In this context it is noteworthy that, in the IR measurements of  $\text{NH}_3$  adsorption on  $\text{Ru}(0001)$  by Rodriguez et al.,<sup>19</sup> only a single feature for chemisorbed  $\text{NH}_3$  is observed.

The adsorption of  $\text{NH}_3$  is completely reversible for both  $\text{Pd}_{1\text{ML}}/\text{Mo}(110)$  and  $\text{Pd}(111)/\text{Mo}(110)$ ; no indication of dissociation is found in the TPD and HREELS data. The adsorption of  $\text{NH}_3$  on several of the later transition metals has also been found previously to be largely reversible. These include  $\text{Ru}(001)$ ,  $\text{Pt}(111)$ ,  $\text{Ni}(111)$ , and  $\text{Ni}(110)$ . Only on Ni–

(110) has a substantial amount of dissociation been found after dosing  $\text{NH}_3$  at room temperature.<sup>44</sup>

HREELS has been used to study  $\text{NH}_3$  adsorption on Ru(001), Pt(111), Ni(111), Ni(110), Ag(110), Ag(311), and Fe(110). The HREEL spectra of chemisorbed  $\text{NH}_3$  in most of these cases are dominated by the  $\text{NH}_3$  asymmetric deformational mode. The wavenumber of this vibrational feature has been correlated either to the strength of the  $\text{NH}_3$ –metal bond<sup>39</sup> or to the charge density at the substrate atoms.<sup>20</sup>  $\text{NH}_3$  adsorbed on NiO(100) shows a very similar TPD spectra as for the transition metal; however, no shift in the  $\text{NH}_3$  asymmetric deformational mode with increasing  $\text{NH}_3$  coverage is apparent. This is likely due to the localized nature of the charge in an insulating oxide. Thus, the shift of the symmetric deformational mode with increasing  $\text{NH}_3$  coverage is likely dominated by effects related to the electron density at the surface. In any event, this shift is a measure of the ability of the substrate to accept and stabilize the nitrogen electron lone pair of  $\text{NH}_3$ . This, in turn, correlates to the density of states near the Fermi level as well as the electron density within individual orbitals. That this vibration has exactly the same frequency for both  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  and  $\text{Pd}(111)/\text{Mo}(110)$  surfaces at lower  $\text{NH}_3$  coverage and decreases to lower frequency at a more rapid rate on the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface than on the  $\text{Pd}(111)/\text{Mo}(110)$  surface indicates a slightly enhanced reduction in the density of states near the Fermi level for the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface compared to  $\text{Pd}(111)/\text{Mo}(110)$ , but a very similar electron density of individual orbitals for both systems. In this respect a slightly reduced high-temperature state in TPD for  $\text{NH}_3$  adsorbed on  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  in comparison to the  $\text{Pd}(111)/\text{Mo}(110)$  surface has been found.

Previously, a reduced binding energy has been found for CO bound to a Pd monolayer on a number of substrates, including transition metals, noble metals, and sp metals such as Al. This indicates that the reduced reactivity of monolayer Pd is an intrinsic property. An interaction with the substrate will certainly further modify the electronic structure of the Pd monolayer, which accounts for the differences in Pd monolayers supported on various substrates since the CO reduced binding energy has been attributed primarily to reduced back-bonding from the metal. In this study, the reduced ability of a Pd monolayer to accept electrons has been demonstrated and is clearly due to a reduced density of states near the Fermi level. The monolayer presents a transition from an atomiclike structure with discrete states to a band structure with continuous states. Angular-resolved ultraviolet photoemission spectroscopy (ARUPS) of monolayer Pd on Al(111), where the features corresponding to the Pd monolayer can be clearly distinguished from the substrate emission, shows that the transition from the atomiclike structure is not yet complete at a monolayer coverage of Pd. That is, an atomiclike electronic structure in a Pd monolayer is retained to a degree. Consequently, the density of states near the Fermi level for the monolayer is less than bulk Pd, and the monolayer shows reduced reactivity.

**4.2. Coadsorption of  $\text{NH}_3$  with CO.** Coadsorption of CO and  $\text{NH}_3$  causes a blue shift of the  $\text{NH}_3$  symmetric deformation mode and a red shift of the C–O stretching mode for monolayer and multilayer Pd. However, each of these shifts occurs to a significantly larger extent for  $\text{Pd}(111)/\text{Mo}(110)$  than  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$ . This is consistent with a higher binding energy for CO and  $\text{NH}_3$  on  $\text{Pd}(111)/\text{Mo}(110)$  compared to  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  and indicates that the interaction between CO and  $\text{NH}_3$  occurs primarily through the surface.

The frequency shift to lower wavenumber of the C–O vibrational mode induced by a coadsorbed electropositive

species such as ammonia or alkali metals has been a subject of intensive debate. Several mechanisms have been proposed to explain this red shift. In the chemical shift model, the coadsorption of an electron donor leads to greater surface electron density, therefore leading to an enhanced electron transfer into the  $2\pi^*$  antibonding orbital of the adsorbed CO. This occurs through either depolarization in the case of a more localized electron donor such as  $\text{NH}_3$  or electron transfer into the substrate in the case of alkali metals. The consequence of the enhanced back-bonding is that the C–O bond is weakened while the CO–surface bond is stabilized. Concomitantly, the C–O stretching mode shifts to lower wavenumber. An increased binding energy of CO is also observed with the coadsorption of alkali metals. Since  $\text{NH}_3$  desorbs well below CO from transition metals, the influence of coadsorbed  $\text{NH}_3$  on the CO binding energy cannot be addressed using conventional TPD. However, the above arguments suggest that CO coadsorbed with  $\text{NH}_3$  will have a higher binding energy. In contrast, the electrostatic model argues that adsorbed CO and  $\text{NH}_3$  have opposite dipoles with respect to the surface normal. The dipole–dipole interaction then stabilizes both molecules on the surface. The blue shift is simply a result of the interaction of CO with the electrostatic field produced by the coadsorbed molecules (Stark shift).

In actuality, a combination of both charge transfer and electrostatic interaction exists. To separate these effects, Hoffman et al. studied coadsorption of CO with Xe on the Ru(001) surface.<sup>47</sup> Since Xe is only weakly physisorbed and charge transfer is negligible, the observed red shift of the C–O vibrational mode can be assumed to be due solely to an electrostatic interaction. Yates and co-workers<sup>48–50</sup> have recently studied coadsorption of CO with Xe,  $\text{H}_2$ , or O and were able to correlate the shift of the C–O vibrational mode with the global work function change. These authors argued that the shift of the C–O vibrational mode was primarily due to the alteration of the Fermi level by the coadsorbate. Depending upon the coadsorbed species, the Fermi level moves either up or down, resulting in an increase or decrease in the back-bonding and, therefore, a red or blue shift of the C–O stretching mode. The dipole–dipole interaction plays only a minor role in the observed frequency shift.

The present results for  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  and  $\text{Pd}(111)/\text{Mo}(110)$  are consistent with both long-range and short-range interactions. The long-range interaction, likely due to a work function change, is most clearly seen at lower CO coverages on  $\text{Pd}(111)/\text{Mo}(110)$ . At 0.05 langmuir CO exposure, a gradual shift of both the C–O stretching mode and the  $\text{NH}_3$  symmetric deformation to higher frequencies with decreasing  $\text{NH}_3$  coverage is observed. Desorbing  $\text{NH}_3$  causes a gradual increase in the work function and therefore a decrease in the  $d-\pi^*$  back-bonding. The  $\text{NH}_3$  symmetric deformational mode also shifts to higher frequency due to an attenuated  $\text{NH}_3$ – $\text{NH}_3$  interaction. This result can also be rationalized in terms of the work function change, which enhances the ability of the substrate to accept electrons from  $\text{NH}_3$ , thus forming a more stable bond. In any event, it is clearly a long-range, rather than a short-range, effect. On the  $\text{Pd}_{\text{IML}}/\text{Mo}(110)$  surface, the interaction between  $\text{NH}_3$  and CO also results in a change in the adsorption site at relatively high annealing temperatures, from the atop site to a likely 3-fold hollow site. With respect to the 3-fold hollow site, the C–O vibrational mode gradually shifts from 1680  $\text{cm}^{-1}$  at higher CO exposures to 1810  $\text{cm}^{-1}$  at a CO exposure of 0.05 langmuir. This shift (130  $\text{cm}^{-1}$ ) is somewhere smaller than the corresponding shift (200  $\text{cm}^{-1}$ ) found for the 3-fold hollow site on



the Pd(111)/Mo(110) surface. This is consistent with CO and NH<sub>3</sub> being more strongly bonded to the Pd(111)/Mo(110) surface compared to the Pd<sub>1ML</sub>/Mo(110) surface. The interaction through the substrate is also more significant.

In addition to long-range interactions, short-range interactions are also clearly evident. For instance, the dosing of NH<sub>3</sub> on top of a CO-covered surface converts a single C—O vibration feature to two C—O vibrational features or occasionally a broad peak with a foot extending to lower wavenumbers. These results suggest either a different number of NH<sub>3</sub> neighboring molecules of CO or a reduced density in the NH<sub>3</sub> neighbors. This short-range effect is probably dominated by local charge transfer and, to a lesser degree, dipole–dipole interactions. In a charge-transfer mechanism, NH<sub>3</sub> donates electron density via the nitrogen atom to the substrate, polarizing the substrate electron density such that the d– $\pi$  back-donation is increased.

It is also noteworthy that the presence of NH<sub>3</sub> promotes the occupancy by CO of more highly coordinated sites on the surface, with the Pd monolayer appearing more bulklike with respect to CO adsorption. It has been demonstrated<sup>18</sup> that CO bonded to a 3-fold hollow site is more strongly destabilized relative to the atop site in going from bulk Pd to monolayer Pd. As a result, the atop site is the most stable adsorption site for CO on the supported Pd monolayer, while the three-hollow site is the most stable adsorption site for CO on bulk Pd. The presence of the NH<sub>3</sub> molecule, an electron donor, on the Pd<sub>1ML</sub>/Mo(110) surface partially restores the adsorption behavior of bulk Pd with respect to CO. This again is due to electron donation from NH<sub>3</sub> to the surface and implies that the availability of electron density for back-donation in CO bonding to a transition-metal surface is the dominating factor in determining the adsorption site of CO.

Since both CO–CO and NH<sub>3</sub>–NH<sub>3</sub> interactions are repulsive and the CO–NH<sub>3</sub> interaction is attractive, coadsorbed CO and NH<sub>3</sub> form mixed, rather than separate, domains. Even at the low coverage limit, a change in the bonding site of CO from atop to the 3-fold hollow, induced by coadsorbed NH<sub>3</sub> molecules, is consistent with the formation of a (CO)<sup>8-</sup>–(NH<sub>3</sub>)<sup>8+</sup> complex due to the attractive interaction between CO and NH<sub>3</sub>.

## 5. Conclusion

Using NH<sub>3</sub> as a probe, the Pd monolayer has been shown to have a slightly reduced density of states near the Fermi level and thus is better able to accept electron density from NH<sub>3</sub> compared with the bulk-terminated Pd surface. These results together with previous results obtained for these systems show that monolayer Pd compared with bulk Pd has a more deficient density of state near the Fermi level. The results of NH<sub>3</sub> and CO coadsorption show an attractive interaction between these two molecules, a significantly modified characteristic vibrational mode, and a change in the CO adsorption site on the Pd<sub>1ML</sub>/Mo(110) surface.

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## References and Notes

- (1) Rodriguez, J. A. *Surf. Sci. Rep.* **1996**, 24, 223.
- (2) Bauer, E. *The Chemical Physics of Solid Surfaces and Heterogeneous Catalysis*; King, D. A., Woodruff, D. P., Eds.; Elsevier: Amsterdam, 1984.
- (3) Campbell, C. T. *Annu. Rev. Phys. Chem.* **1990**, 41, 775.
- (4) Rodriguez, J. A.; Goodman, D. W. *Science* **1992**, 257, 897.
- (5) Rodriguez, J. A.; Campbell, R. A.; Goodman, D. W. *Surf. Sci.* **1994**, 377, 307.
- (6) Weng, S.-L.; El-Batanouny, M. *Phys. Rev. Lett.* **1980**, 44, 612.
- (7) El-Batanouny, M.; Strongin, M.; Williams, G. P.; Colbert, J. *Phys. Rev. Lett.* **1981**, 46, 269.
- (8) Williams, G. P.; El-Batanouny, M.; Colbert, J.; Jensen, E.; Rhodin, T. N. *Phys. Rev.* **1982**, B25, 3658.
- (9) Ruckman, M. W.; Johnson, P. D.; Strongin, M. *Phys. Rev.* **1985**, B31, 3405.
- (10) Frick, B.; Jacobi, K. *Phys. Rev. B* **1988**, 37, 4408.
- (11) Koel, B. E.; Smith, R.; Berlowitz, P. J. *Surf. Sci.* **1990**, 231, 325.
- (12) Campbell, R.; Rodriguez, J.; Goodman, D. W. *Surf. Sci.* **1990**, 240, 71.
- (13) Heitzinger, J. M.; Gebhard, S. C.; Koel, B. E. *Surf. Sci.* **1992**, 275, 209.
- (14) Heitzinger, J. M.; Gebhard, S. C.; Koel, B. E. *Chem. Phys. Lett.* **1992**, 200, 65.
- (15) Sellidj, A.; Koel, B. E. *Surf. Sci.* **1993**, 281, 223.
- (16) Heitzinger, J. M.; Gebhard, S. C.; Koel, B. E. *J. Phys. Chem.* **1993**, 97, 5327.
- (17) Kuhn, W. K.; Szanyi, J.; Goodman, D. W. *Surf. Sci.* **1994**, 303, 377.
- (18) Xu, C.; Goodman, D. W. *Surf. Sci.* **1996**, 360, 249.
- (19) Rodriguez, J. A.; Kuhn, W. K.; Truong, C. M.; Goodman, D. W. *Surf. Sci.* **1992**, 271, 333.
- (20) Parmeter, J. E.; Wang, Y.; Mullins, C. B.; Weinberg, W. H. *J. Chem. Phys.* **1988**, 88, 5225.
- (21) Erley, W.; Ibach, H. *Surf. Sci.* **1982**, 119, L357.
- (22) Gland, G. L.; Kollin, E. B. *Surf. Sci.* **1981**, 104, 478.
- (23) Sexton, B. A.; Mitchell, G. E. *Surf. Sci.* **1980**, 99, 523.
- (24) Sexton, B. A.; Mitchell, G. E. *Surf. Sci.* **1980**, 99, 539.
- (25) Seabury, C. W.; Rhodin, T. N.; Purtell, R. J.; Merrill, R. P. *Surf. Sci.* **1980**, 93, 117.
- (26) Netzer, F. P.; Madey, T. E. *Phys. Rev. Lett.* **1981**, 47, 928.
- (27) Netzer, F. P.; Madey, T. E. *Surf. Sci.* **1982**, 119, 422.
- (28) Kang, W. M.; Li, C. H.; Tang, S. Y.; Seabury, C. W.; Jacobi, K.; Rhodin, T. N.; Purtell, R. J.; Merrill, R. P. *Phys. Rev. Lett.* **1981**, 47, 931.
- (29) Grunze, M.; Golze, M.; Driscoll, R. K.; Dowber, P. A. *J. Vac. Sci. Technol.* **1981**, 18, 611.
- (30) Jacobi, K.; Jensen, E. S.; Rhodin, T. N.; Merrill, R. P. *Surf. Sci.* **1981**, 108, 397.
- (31) Gland, J. L.; Sexton, B. A.; Mitchell, G. E. *Surf. Sci.* **1982**, 115, 623.
- (32) Ceyer, S. T.; Yates, J. T. *Surf. Sci.* **1985**, 155, 584.
- (33) Netzer, F. P.; Madey, T. E. *Chem. Rev. Lett.* **1982**, 88, 315.
- (34) Wu, M. C.; Truong, C. M.; Goodman, D. W. *J. Phys. Chem.* **1993**, 97, 4182.
- (35) Tochiwara, H.; Rocker, G.; Redding, J. D.; Yates, Jr., J. T.; Martin, R. M.; Metiu, H. *Surf. Sci.* **1986**, 176, 1.
- (36) Dresser, M. J.; Lanzillotto, A. M.; Alvey, M. D.; Yates, Jr., J. T. *Surf. Sci.* **1987**, 191, 1.
- (37) Lanzillotto, A. M.; Dresser, M. J.; Alvey, M. D.; Yates, Jr., J. T. *Surf. Sci.* **1987**, 191, 15.
- (38) Erley, W. *Surf. Sci.* **1990**, 240, 1.
- (39) Zhou, Y.; Akhter, S.; White, J. M. *Surf. Sci.* **1988**, 202, 357.
- (40) Sasaki, T.; Aruga, T.; Kuroda, H.; Iwasawa, Y. *Surf. Sci.* **1989**, 224, L969.
- (41) Rosenzweig, Z.; Asscher, M.; Wittenzellner, C. *Surf. Sci. Lett.* **1990**, 240, 583.
- (42) Lackey, D.; King, D. A. *J. Chem. Soc., Faraday Trans.* **1987**, 83, 2001.
- (43) Mijoule, C.; Bouteiller, Y.; Salahub, D. R. *Surf. Sci.* **1991**, 253, 375.
- (44) Parmeter, J. E.; Jiang, X.; Goodman, D. W. *Surf. Sci.* **1990**, 240, 85.
- (45) Parker, Ch.; Bauer, E.; Poppa, H. *Surf. Sci.* **1985**, 154, 371.
- (46) Benndorf, C.; Madey, T. E. *Surf. Sci.* **1983**, 135, 164.
- (47) Hoffmann, F. M.; Lang, N. D.; Nørskov, J. K. *Surf. Sci. Lett.* **1990**, 226, L48.
- (48) Xu, Z.; Sherman, M. G.; Yates, Jr., J. T.; Antoniewicz, P. R. *Surf. Sci.* **1992**, 276, 249.
- (49) Zenobi, R.; Xu, J.; Yates, Jr., J. T. *Surf. Sci.* **1992**, 276, 241.
- (50) Xu, Z.; Surnev, L.; Uram, K. J.; Yates, Jr., J. T. *Surf. Sci.* **1993**, 292, 235.