

ARTICLES

Direct Cross-Linking of C₇₀ in Ar Plasma

Masafumi Ata,* Ken'ichi Kurihara, and Noboru Takahashi

*Hodogaya Laboratory, Research Institute of Innovative Technology for the Earth, Sony Corporation Research Center, 174 Fujitsuka-cho, Hodogaya-ku, Yokohama 240, Japan**Received: March 4, 1996; In Final Form: September 4, 1996*[®]

C₇₀ polymer films were prepared by rf plasma irradiation in Ar atmosphere. Laser desorption ionization time-of-flight mass spectra (LDITOF-MS) for the polymer films showed that C₂ loss occurred in the course of direct cross-linking of C₇₀ molecules. The polymer structure is discussed based on molecular orbital (MO) calculations of C₇₀ dimers at semiempirical levels, and the lower probability of direct cross-linking of C₇₀ molecules than that of C₆₀ molecules is suggested. The electric dark current of the plasma-polymerized C₇₀ was 10⁻¹³ S/cm, compared to 10⁻⁸ S/cm in the C₆₀ polymer prepared at the same plasma power. The almost-insulating electric conductivity of the C₇₀ polymer is attributable to the lower probability of conductive cross-linking of C₇₀ molecules originating from the electronic structure of C₇₀. The experimental and theoretical results suggest that the C₇₀ molecule is the smallest carbon nanotube with polyhedral caps and that the chemical properties of the caps closely resemble the C₆₀ molecule.

Introduction

Discovery of buckminsterfullerene (C₆₀)¹ and a novel method for preparing macroscopic amounts of fullerenes² has led to a significant volume of research on its unique physical and chemical properties. Polymers of C₆₀ have been obtained by direct photochemical cross-linking of C₆₀ molecules.³ For the photoinduced C₆₀ polymer, the mechanism for the photopolymerization of solid C₆₀,⁴ the kinetics of the photodimerization process,⁵ photoelectron spectroscopy,⁶ and Raman spectra⁷ have been reported. Odd-numbered carbon clusters derived from C₆₀ in an oxygenated atmosphere have been reported by McElvany *et al.*⁸ Recently, linear polymer of C₆₀ was confirmed in K- and Rb-doped C₆₀ crystal.⁹ Iwasa *et al.*¹⁰ reported a phase transition of C₆₀ crystal at high pressure and proposed a novel two-dimensional C₆₀ polymer,¹¹ whose properties were theoretically examined.¹²

To obtain C₆₀ polymer, we have adopted a plasma-polymerization technique.¹³ The electric dark current of the polymer films was ca. 10⁻⁸ S/cm in ambient atmosphere, and a semiconductor-type temperature dependence of the conductivity at high temperatures domain was observed, suggesting a bandgap energy of 2.1 eV.¹³ On the basis of MO calculations we proposed that the 1,2-C₆₀ dimer molecule by [2+2]-cycloaddition was found to be the most plausible structure of a C₆₀ dimer with intact C₆₀ cages.¹⁴ Similar theoretical results have been reported by several groups.¹⁵ To further clarify the polymer structure, LDITOF and field desorption mass spectra of plasma-polymerized C₆₀ films were measured and the dimer peaks were assigned with the aid of MO calculations.¹⁶ There is evidence that four carbon atoms are frequently eliminated in the cross-linking of two C₆₀ molecules. The most intense peak in the dimer mass range was therefore assigned to a C₁₁₆ molecule with D_{2h} symmetry.¹⁶

Whereas many derivatives of C₆₀ have been investigated, only a few derivatives of C₇₀,¹⁷ metal complexes of C₇₀,¹⁸ alkylated

C₇₀,¹⁹ and hydrated and halogenated C₇₀²⁰ have been isolated and characterized. Photopolymerization of C₇₀ was examined by Rao *et al.*²¹ Their findings indicated that the molecule is less active than C₆₀ toward cycloaddition.

In this article, we report the plasma polymerization of C₇₀ in an Ar atmosphere. Raman spectra of polymer films of C₇₀ are compared with the spectra of the pure C₇₀ cage²² and amorphous carbon films.²³ Furthermore, LDITOF mass analyses of the C₇₀ polymer films are reported. Based on these observations and MO calculations at semiempirical levels, the structure of the polymer and the polymerization process as well as the mechanism under Ar plasma irradiation are discussed. Finally, the polymerization process of C₇₀ is compared with that of C₆₀, and the characteristics of C₇₀ polymers are presented.

Experimental Section

The C₇₀ sample was purified using an activated aluminum column 200 cm long, with a *n*-hexane:toluene 9:1 v/v solution. The C₇₀ sample was sealed in an evacuated glass tube after being heated for 12 h at 300 °C, and the sample was then crystallized using an electric heater equipped with two heating probes, which were set at 590 and 500 °C. The preparation of the plasma-polymerized C₇₀ films was carried out in an Ar plasma at a power of 50 W. The other experimental details of plasma polymerization were identical to those reported previously.¹³

LDITOF-MS observations were carried out using JEOL LDI-1700 and Finnigan MAT Vision 2000 mass monitors, and both sets of apparatus were equipped with a N₂ laser. Mass monitoring of the plasma-polymerized C₇₀ films was carried out without any matrix molecule for the reasons stated previously.¹⁶ To compare the probability of direct cross-linking of C₆₀ and C₇₀ by UV irradiation, samples of C₆₀, C₇₀, and a 1:1 mixture of C₆₀/C₇₀ were set on ionization targets, and the polymer peaks were observed. During the observation, the laser power was increased sufficiently to cause photopolymerization.

An Ar⁺ laser (514.5 nm) used for excitation in Raman observation was operated at 20 mW and the beam spot was 1

[®] Abstract published in *Advance ACS Abstracts*, December 15, 1996.

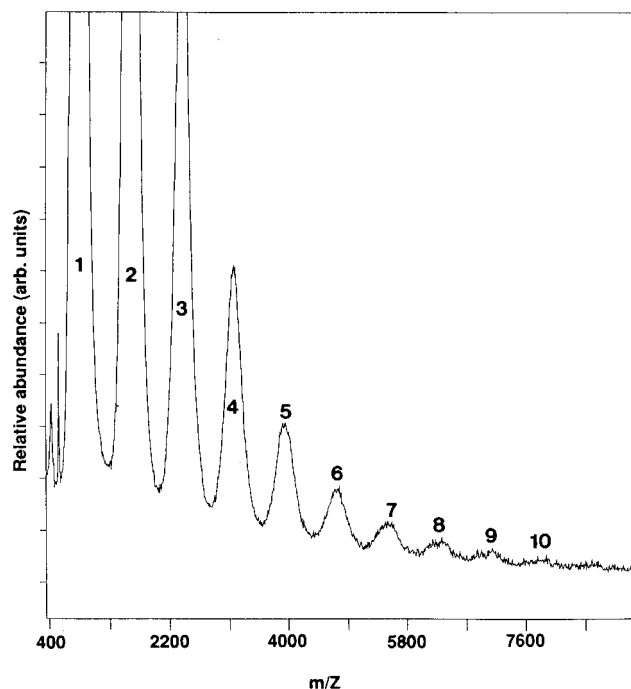


Figure 1. LDITOF-MS of a plasma-polymerized C_{70} film observed at the laser-desorption power of $2.38 \mu\text{J}$ per laser pulse. The spectrum was obtained without a matrix molecule such as cinapinic acid to assist the desorption and ionization. Thus, the peaks so obtained are free from H^+ adduct of the sample, though positive ions were detected in the actual measurements.

mm in diameter. Raman measurements were performed in a back-scattering geometry at room temperature using a JOBIN YBON T64000 triple monochromator equipped with a two-dimensional CCD detector operated at liquid-nitrogen temperature.

Calculations

Dimers of C_{70} were calculated using the MNDO Hamiltonian²⁴ with the AM-1 and PM-3 parametrizations²⁵ as implemented in the MOPAC program system.²⁶ For ease of direct comparison with literature data,^{13,16} version 5.01 was used in this study. Symmetry constraints were not specified for the geometry optimization procedure, and the SCF convergence criteria were set at 10^{-8} kcal/mol.

A conventional [2+2]cycloaddition reaction²⁷ of C_{70} molecules was considered to be the initial stage of cross-linking, based on the theoretical results for C_{60} dimers.^{13,16} Calculations were limited to the dimer structures of C_{70} having symmetries higher than C_{2v} and C_{2h} . Prior to the MO calculations, structures of the C_{70} dimers were optimized using molecular mechanics (MM) calculations.²⁸

Results

Figure 1 shows the LDITOF-MS of the C_{70} polymer film observed at laser power of $2.38 \mu\text{J}$. At the same laser power, LDITOF-MS of an evaporated C_{70} film showed that the proportion of dimer to monomer was less than 1:60. Thus, Figure 1 confirms the plasma polymerization of C_{70} molecules by Ar plasma irradiation prior to the measurements, rather than by photoradiation during LDITOF-MS observation. It can be seen in Figure 1 that the maximum mass values of the polymer peaks do not coincide with multiples of the initial mass of C_{70} . For example, the fifth peak was observed at about 4000 Da, which does not correspond to the mass of $(C_{70})_5$ ($\text{MW} = 4204$). Thus, direct cross-linking of C_{70} molecules must be accompanied by the loss of carbon atoms during plasma polymerization. This was confirmed by measuring the LDITOF-MS of the plasma-

polymerized samples in the monomer and dimer mass regions at a slightly reduced laser power.

Figure 2a, b, and c shows the peak profiles of the monomer and dimer mass region of the C_{70} polymer film. The monomer envelope is composed of peaks associated with C_{70} , C_{68} , C_{66} , C_{64} , C_{62} , C_{60} , and C_{58} showing the C_2 loss during plasma irradiation. The dimer profile shown in Figure 2b is composed of strong peaks corresponding to $(C_{70})_2$, $(C_{70})_2 - (\text{C}_2)_n$ ($n = 1-5$), and very weak peaks corresponding to $(C_{70})_2 + (\text{C}_2)_n$ ($n = 1-2$). The higher peaks are C_{136} and C_{134} , and the peak of $(C_{70})_2$ is low. This peak profile closely resembles that of photopolymerized C_{60} ^{5,6} and that of plasma-polymerized C_{60} .¹⁶ Figure 2c shows the LDITOF-MS of a C_{70} polymer generated under reduced plasma power at 25 W. The highest peak is C_{136} in this case, and the spectrum suggests that the degree of $(\text{C}_2)_n$ loss is dependent on the plasma power.

Raman spectra of the plasma-polymerized C_{60} and C_{70} films are shown in Figure 3, a and b, respectively. The films are photoemissive as can be seen in the spectra, though the emission is weak at ambient temperature. Raman shifts attributable to intact C_{60} and C_{70} cages were scarcely observed in the spectra. However, it is clear that the amount of intact cages remaining in the plasma-polymerized films is higher for C_{70} than for C_{60} . The spectral profile suggests that the plasma-polymerized C_{70} film is a carbon film with sp^2 carbons since the graphitic band appeared at 1570 cm^{-1} . A disordered band due to amorphous carbon, which usually appears at 1350 cm^{-1} , was scarcely observed in this case. X-ray diffraction measurements of the C_{70} polymer film formed on a glass substrate 300 nm in thickness showed no diffraction pattern, suggesting the lack of periodic crystal structure in the film.

The electric dark current of the C_{70} polymer formed on an Au-comb electrode was about 10^{-13} S/cm , indicating that the C_{70} polymer was almost insulating. Scanning electron microscope images of the plasma-polymerized C_{70} showed a hemispherical morphology, as observed for the plasma-polymerized C_{60} .¹³ Finally, electron spin resonance measurement of the plasma-polymerized C_{70} film showed that the number of dangling bonds remained on the same level as found for plasma-polymerized C_{60} film.¹³

Discussion

Our examination of the structure of C_{60} dimers having dumbbell shapes with intact C_{60} cages formed by 1,2-, 1,4-, and 1,2+1,4-cycloaddition of a cyclohexatrienyl unit to each of the C_{60} molecules suggested that the 1,2-cycloaddition reaction led to the most stable dimer.¹⁴ Initially, the theoretical examination of C_{70} dimers produced by plasma polymerization was limited to those formed by the [2+2]cycloaddition reaction. Figure 4 shows the optimized structure of C_{70} and the numbering system adopted in this work. The 105 C-C bonds in a C_{70} molecule having D_{5h} symmetry can be classified into eight types, as shown in Table 1. The C-C bonds between two adjacent hexagons are double bond in nature except for the C(11)-C(12) bond, which seems to be a single bond. It is noteworthy that the bond length and bond order of the C(9)-C(10) bond are double-bond-like, although the bond participates in the formation of a pentagon. We chose the C(2)-C(4), C(5)-C(6), C(9)-C(10), C(10)-C(11), and C(11)-C(12) bonds to examine [2+2]-cycloaddition of the C_{70} molecules. The C(11)-C(12) bond is included in the examination since this bond is 6,6-ring fusion. Twenty-five structures of C_{70} dimers are possible by [2+2]-cycloaddition of these bonds, among which only nine C_{70} dimers formed by [2+2]cycloaddition of identical bonds of two C_{70} molecules were examined.

The cycloaddition of C(2)-C(4), C(5)-C(6), C(9)-C(10), and C(10)-C(11) bonds leads to structural isomers, while that

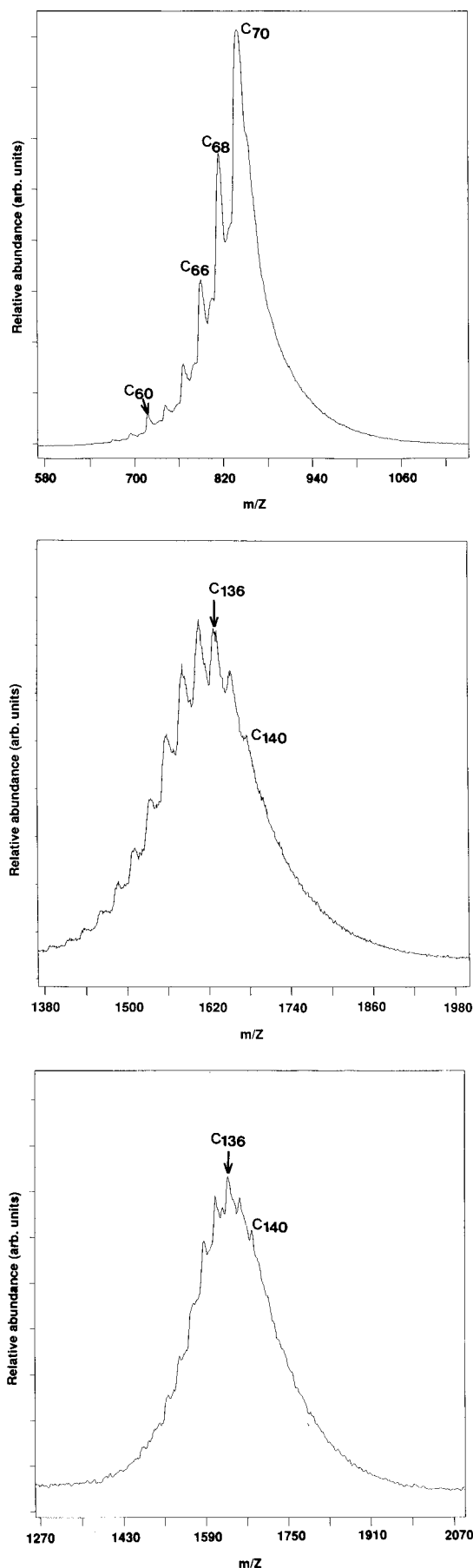


Figure 2. LDITOF-MS of the monomer (top) and dimer (middle) mass range observed at lower laser-desorption power of 1.20 μ J per laser pulse repeated 50 times. The bottom spectrum is dimer mass range observed on the C₇₀ polymer film prepared at a plasma power of 25 W.

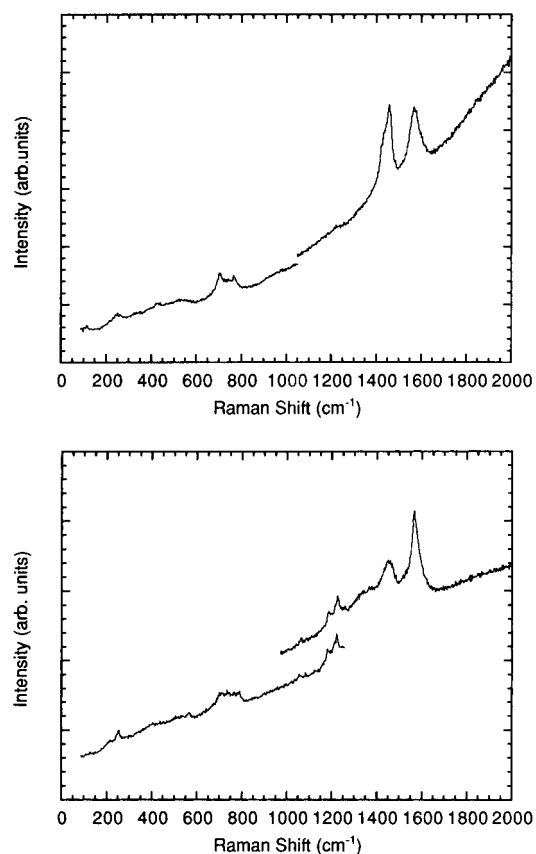


Figure 3. Raman spectra of the plasma-polymerized C₆₀ (top) and C₇₀ films (bottom). The 514.5 nm line from the Ar⁺ laser was used for excitation.

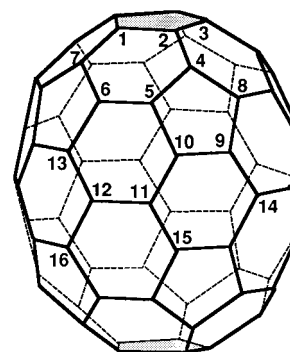


Figure 4. Optimized C₇₀ structure and the numbering system adopted.

TABLE 1: Optimized C–C Bond Lengths and Bond Orders of C₇₀ at MNDO/AM-1 Level^a

bond ^b	bond length/Å	bond order
C(1)–C(2)	1.464	1.105
C(2)–C(4)	1.387	1.480
C(4)–C(5)	1.461	1.109
C(5)–C(6)	1.375	1.526
C(5)–C(10)	1.467	1.065
C(9)–C(10)	1.434	1.302
C(10)–C(11)	1.414	1.306
C(11)–C(12)	1.465	1.094

^a The two types of C–C bond lengths in C₆₀ molecule were calculated as 1.385 and 1.464 Å, and bond orders were calculated as 1.495 and 1.101 for each bond, respectively. ^b Numbering system adopted is shown in Figure 4.

of C(11)–C(12) bonds results in a structure with D_{2h} symmetry. The heats of reaction ($\Delta H_f^\circ(r)$) in the formation of C₇₀ dimers are listed in Table 2. The difference in kinetic energy between the anti-syn isomers is negligible for all pairs formed by cycloaddition of C(2)–C(4), C(5)–C(6), C(9)–C(10), and

TABLE 2: MNDO Heats of Reaction ($\Delta H^\circ(r)$) in the Formation of C_{70} Dimers with Cyclobutane Ring

cluster	$\Delta H_f^\circ(r)^a$ (kcal/mol)	$\Delta H_f^\circ(r)^b$ (kcal/mol)	cross-link ^c	length ^d (Å)
C ₁₄₀ (a)	-34.63	-38.01	C(2)–C(2'), C(4)–C(4')	1.544
			C(2)–C(4), C(2')–C(4')	1.607
C ₁₄₀ (b)	-34.33	-38.00	C(2)–C(4'), C(4)–C(2')	1.544
			C(2)–C(4), C(2')–C(4')	1.607
C ₁₄₀ (c)	-33.94	-38.12	C(5)–C(5'), C(6)–C(6')	1.550
			C(5)–C(6), C(5')–C(6')	1.613
C ₁₄₀ (d)	-33.92	-38.08	C(5)–C(6'), C(6)–C(5')	1.551
			C(5)–C(6), C(5')–C(6')	1.624
C ₁₄₀ (e)	-19.05	-20.28	C(9)–C(9'), C(10)–C(10')	1.553
			C(9)–C(10), C(9')–C(10')	1.655
C ₁₄₀ (f)	-18.54	-19.72	C(9)–C(10'), C(10)–C(9')	1.555
			C(9)–C(10), C(9')–C(10')	1.655
C ₁₄₀ (g)	+3.19	-3.72	C(10)–C(10'), C(11)–C(11')	1.559
			C(10)–C(11), C(10')–C(11')	1.613
C ₁₄₀ (h)	+3.27	-3.23	C(10)–C(11'), C(11)–C(10')	1.560
			C(10)–C(11), C(10')–C(11')	1.613
C ₁₄₀ (i)	+64.30	+56.38	C(11)–C(11'), C(12)–C(12')	1.560
			C(11)–C(12), C(11')–C(12')	1.683

^a MNDO/AM-1 heat of reaction. ^b MNDO/PM-3 heat of reaction.

^c Numbering system is same as shown in Figure 4. The carbons represented as n' are those of the neighbor C_{70} in the same manner.

^d C–C bond lengths of pivot bonds forming cyclobutane rings predicted at the MNDO/AM-1 level. At the same level, the C–C bond lengths of ethane, benzene, and ethylene are calculated to be 1.500, 1.395, and 1.326 Å, respectively.

C(10)–C(11) bonds. The $\Delta H^\circ(r)$ values of the C_{70} dimers reveal that the stable dimers are derived from the [2+2]-cycloaddition of the C(2)–C(4) bonds and C(5)–C(6) bonds of C_{70} molecules. The $\Delta H^\circ(r)$ values of the formation of C₁₄₀-(a), C₁₄₀(b), C₁₄₀(c), and C₁₄₀(d) are very similar to that of the C₆₀ dimer,^{13,16} suggesting that the chemical nature of the C(2)–C(4) and C(5)–C(6) bonds of the C_{70} molecule is almost the same as that of weakly conjugated C–C double bonds in the C₆₀ molecule. In these four C₁₄₀ structures, the bond alternations of the C_{70} moieties are scarcely changed, as in the case of C₆₀ dimer by [2+2]cycloaddition.¹⁶

The structures C₁₄₀(e) and C₁₄₀(f) are destabilized by 15 and 18 kcal/mol at the AM-1 and PM-3 levels, respectively, compared to the structures C₁₄₀(a), C₁₄₀(b), C₁₄₀(c), and C₁₄₀(d). The isomers of C₁₄₀(g) and C₁₄₀(h) were found to be endothermic at the AM-1 level only, whereas the C₁₄₀(i) is strongly destabilized. The $\Delta H^\circ(r)$ values of the C_{70} dimer formed by cycloaddition of the C(1)–C(2) bonds, whose pivot bonds are C(1)–C(2') and C(1')–C(2), are +0.19 and -1.88 kcal/mol at AM-1 and PM-3 levels, respectively. These values are very similar to those of the formation of C₁₄₀(g) and C₁₄₀(h). Thus, the [2+2]cycloaddition reaction of C_{70} molecules can only occur at the C(2)–C(4), C(5)–C(6), and C(9)–C(10) bonds and the probability of the reaction of the C(9)–C(10) bonds is lower than those of the C(2)–C(4) bonds and C(5)–C(6) bonds. The cycloaddition of C(10)–C(11) and C(11)–C(12) bonds should be dynamically impossible.

To clarify the contribution of the steric strain to $\Delta H^\circ(r)$ brought about by the overlapping $2p_z$ lobes of carbon atoms adjacent to the cross-link bonds in the formation of C₁₄₀(a)–C₁₄₀(i), [2+2]cycloaddition of C_{70} and C₆₀ molecules and hydrogenation of C_{70} were theoretically examined. Table 3 shows the ΔH_f° values in the formation of C_{70} dimers and C_{70} –C₆₀ structures by [2+2]cycloaddition, together with the values in the formation of $C_{70}H_2$. In Table 3, all ΔH_f° values are reported relative to that of the C_{70} dimer cross-linked at the C(5) and C(6) atoms, since the MNDO/PM-3 ΔH_f° value for the C₁₄₀(c) is the smallest among those of C_{70} dimers shown in Table 2. In the case of the C_{70} –C₆₀ and $C_{70}H_2$ structures, the difference in ΔH_f° values depends on the change in the electronic structure of the C_{70} moiety brought about by the

TABLE 3: Relative MNDO Heats of Formation (ΔH_f°) in the formation of C_{70} Dimers and C_{70} –C₆₀ Adducts Derived from [2+2]Cycloaddition, and Those in the Formation of $C_{70}H_2$ (in kcal/mol)

form ^a	2(C_{70})		C_{70} –C ₆₀		$C_{70}H_2$	
	AM-1	PM-3	AM-1	PM-3	AM-1	PM-3
C(2), C(4)	-0.69	+0.11	+2.21	+2.46	+1.03	+1.07
	-0.66	+0.12				
C(5), C(6) ^b	0	0	0	0	0	0
	+0.02	+0.04				
C(9), C(10)	+14.89	+17.84	+7.64	+8.89	+6.88	+7.07
	+15.40	+18.40				
C(10), C(11)	+37.13	+34.40	+18.57	+17.59	+19.88	+18.30
	+37.21	+34.89				
C(11), C(12)	+98.24	+94.50	+50.49	+48.33	+51.77	+48.45

^a Carbon atoms forming cyclobutane rings in C_{70} dimers and C_{70} –C₆₀ adducts, and those connected with H atoms in $C_{70}H_2$. ^b ΔH_f° values are chosen as zero reference.

addition reaction. It can clearly be seen in Table 3 that the difference of the ΔH_f° values in the formation of C_{70} dimers is almost twice as large as that found for the formation of the C_{70} –C₆₀ and $C_{70}H_2$ structures. This suggests that the formation of the C_{70} dimers is scarcely affected by the overlapping of the $2p_z$ lobes of carbon atoms adjacent to the cross-link bonds. Instead, only the structural strain in the cyclobutane rings and the changes in the electronic structure of the C_{70} moieties contribute to the ΔH_f° values of C_{70} dimers, within the framework of MNDO approximation.

As is shown in Figure 2, the cross-linking of C_{70} molecules is accompanied by C₂ loss. Therefore, in order to investigate the direct cross-linking of C_{70} molecules in further detail, C₁₃₆ molecules were theoretically examined. The C₁₃₆ molecules examined were constructed as follows: the four carbon atoms participating in the formation of the cyclobutane rings in the C₁₄₀(a)–C₁₄₀(i) molecules are eliminated, and the two C₆₈ moieties with four dangling bonds are recombined. The optimized structures of C₁₃₆ are shown in Figure 5, and the calculated results are collected in Table 4. In this table, the ΔH_f° values for C₁₃₆(c) are chosen as zero reference in order to compare the degree of stability of these C₁₃₆ molecules. The ΔH_f° values suggest that C₁₃₆(a) and C₁₃₆(b) are effectively stabilized by this type of cross-linking. On the other hand, C₁₃₆-(e), C₁₃₆(f), and C₁₃₆(i) are destabilized. The stability of C₁₃₆-(d), C₁₃₆(g), and C₁₃₆(h) is similar to that of C₁₃₆(c). Among these C₁₃₆ structures, only C₁₃₆(a) and C₁₃₆(b) disturb to a small extent the electronic structure of the intact C_{70} molecule.

[2+2]Cycloaddition should be an initial process of direct cross-linking of C₆₀ molecules, and C₂ loss occurs in the C₆₀ dimer.¹⁶ A possible mechanistic pathway of C₂ loss is schematically shown in Figure 6 for the structure C₁₄₀(a). As shown in Table 2, the C(2)–C(2') crosslink bond of C₁₄₀(a) is 1.544 Å, and the C(2)–C(4) bond forming the cyclobutane ring is 1.607 Å. This suggests that highly strained C–C bonds exist in the C(2)–C(4) bond. If scission of these highly strained C(2)–C(4) and C(2')–C(4') bonds occurs, the C₁₄₀(a) is free from sp³ carbon atoms, (Figure 6b). This C_{70} dimer with only sp² carbon atoms, however, is less stable than the initial dimer formed by [2+2]cycloaddition, as seen in the dimerization of C₆₀.¹⁶ The structure shown in Figure 6b is therefore stabilized by rearrangement without C₂ loss, as shown by Scuseria et al.²⁹ and Osawa³⁰ in the case of the C₆₀ dimer. This Stone–Wales rearrangement³¹ results in the formation of a dimer having ladder-shaped crosslinks, (Figures 6c and d). By elimination of two carbon atoms forming one of the ladder-shaped crosslinks in structure 6c, the C₁₃₈ shown in Figure 6e can be derived. Further elimination of C₂ and recombination of the dangling bonds finally results in the formation of C₁₃₆(a) (Figure 6f).

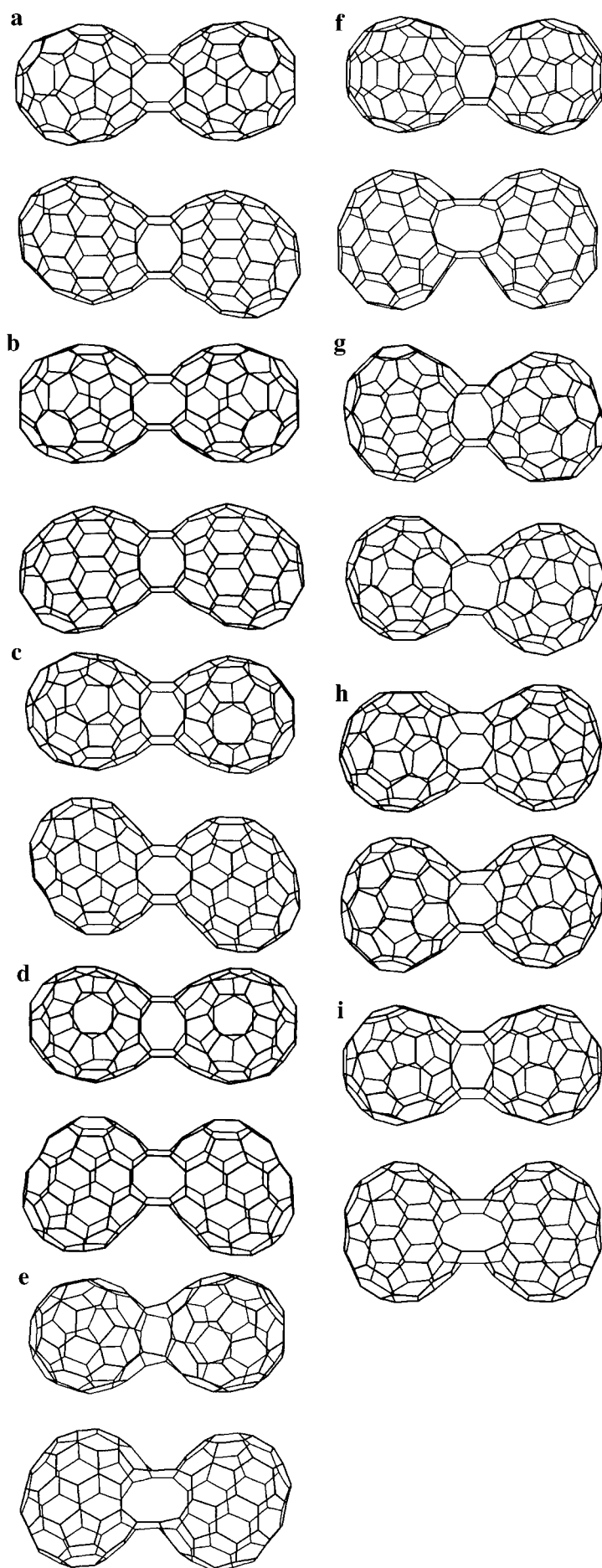


Figure 5. Optimized structures of C_{136} derived from the corresponding C_{70} dimers by elimination of four carbon atoms forming cyclobutane rings and recombination of C_{68} moieties.

TABLE 4: Relative MNDO Heats of Formation (ΔH_f°) for C_{136} Molecules Derived from Corresponding C_{140} Molecules in Table 2

cluster ^a	ΔH_f° ^b (kcal/mol)	ΔH_f° ^c (kcal/mol)	cross-link ^d	length (Å)
$C_{136}(a)$	-65.50	-61.60	C(1)-C(8'), C(3)-C(5')	1.351
			C(5)-C(3'), C(8)-C(1')	1.351
$C_{136}(b)$	-64.44	-61.54	C(1)-C(3'), C(3)-C(1')	1.351
			C(5)-C(8'), C(8)-C(5')	1.351
$C_{136}(c)$	0	0	C(4)-C(13'), C(7)-C(10')	1.352
			C(10)-C(7'), C(13)-C(4')	1.352
$C_{136}(d)$	+0.09	+0.11	C(4)-C(7'), C(7)-C(4')	1.351
			C(10)-C(13'), C(13)-C(10')	1.354
$C_{136}(e)$	+112.98	+102.89	C(5)-C(8'), C(8)-C(5')	1.353
			C(11)-C(14'), C(14)-C(11')	1.372
$C_{136}(f)$	+69.47	+59.44	C(5)-C(14'), C(14)-C(5')	1.358
			C(11)-C(8'), C(8)-C(11')	1.352
$C_{136}(g)$	-3.74	-9.20	C(5)-C(15'), C(15)-C(5')	1.344
			C(12)-C(9'), C(9)-C(12')	1.352
$C_{136}(h)$	+2.82	-5.30	C(5)-C(9'), C(9)-C(5')	1.372
			C(12)-C(15'), C(15)-C(12')	1.334
$C_{136}(i)$	+98.50	+84.36	C(13)-C(10'), C(15)-C(16')	1.376
			C(10)-C(13'), C(16)-C(15')	1.376

^a ΔH_f° values for the structure c are chosen as zero reference.^b MNDO/AM-1 heat of formation. ^c MNDO/PM-3 heat of formation.^d Four pivot bonds in C_{136} clusters.

ΔH_f° values per carbon atom for the above-discussed C_{140} and C_{136} structures are reported in Table 5. At present, we have no clear explanation for the C_2 loss from a highly excited C_{70} molecule as reported in the case of C_{60} .^{32,33} However, from Table 5 it can be seen that the process shown in Figure 6 constitutes a plausible process of structural relaxation of initially formed C_{70} dimer. Recently, Osawa showed that a C_{120} having a one-dimensional tubular shape can be obtained by the Stone–Wales rearrangement from the initial C_{60} dimer,³⁰ while the C_2 loss occurs prior to the type of structural relaxation in the case of plasma polymerization of C_{60} and C_{70} . Not only the C_2 loss but also the addition of C_2 fragments with the C_{70} polymer is

TABLE 5: MNDO Heats of Formation (ΔH_f°) per Carbon Atom for C_{140} and C_{136} Molecules

cluster	$\Delta H_f^\circ/C^a$ (kcal/mol)	$\Delta H_f^\circ/C^b$ (kcal/mol)	cluster	$\Delta H_f^\circ/C^a$ (kcal/mol)	$\Delta H_f^\circ/C^b$ (kcal/mol)
C_{70}	15.171	12.631			
$C_{140}(a)$	14.924	12.359	$C_{136}(a)$	14.643	12.295
$C_{140}(b)$	14.924	12.360	$C_{136}(b)$	14.643	12.295
$C_{140}(c)$	14.929	12.359	$C_{136}(c)$	15.117	12.748
$C_{140}(d)$	14.929	12.359	$C_{136}(d)$	15.118	12.748
$C_{140}(e)$	15.035	12.486	$C_{136}(e)$	15.947	13.546
$C_{140}(f)$	15.039	12.490	$C_{136}(f)$	15.627	13.252
$C_{140}(g)$	15.194	12.607	$C_{136}(g)$	15.090	12.680
$C_{140}(h)$	15.194	12.608	$C_{136}(h)$	15.138	12.709
$C_{140}(i)$	15.630	13.034	$C_{136}(i)$	15.841	13.368

^a MNDO/AM-1 heat of formation per carbon atom. ^b MNDO/PM-3 heat of formation per carbon atom.

stimulated in the course of plasma polymerization of C_{70} . Just as the radical addition reported by Krusic et al.^{34,35} and McEwen et al.,³⁶ the highly active C_2 fragment should be added to the weakly conjugated double bonds of the C_{70} molecule or its polymer in the course of plasma polymerization, forming a cyclobutene structure.

In the above discussion, it was suggested that C_{70} is less reactive than C_{60} toward cycloaddition reaction. To confirm the lower probability of a direct cross-linking of C_{70} molecules, the efficiency of photopolymerization of C_{60} , a 1:1 mixture of C_{60}/C_{70} , and C_{70} was examined using a reflectron type LDITOF-MS monitor. In the observations, the fine crystals of these samples were set on the ionization target and ionized directly by N_2 laser irradiation. The laser power was increased sufficiently to cause photo-polymerization. Figure 7, a, b, and c shows the mass profiles of photopolymerized C_{60} , a C_{60}/C_{70} mixture, and C_{70} , respectively. In this type of photopolymerization, the probability of a direct cross-linking depends on the molecular orientation at the moment of pulse laser irradiation whose pulse width is about 5 ns. As can be seen in Figure 7,

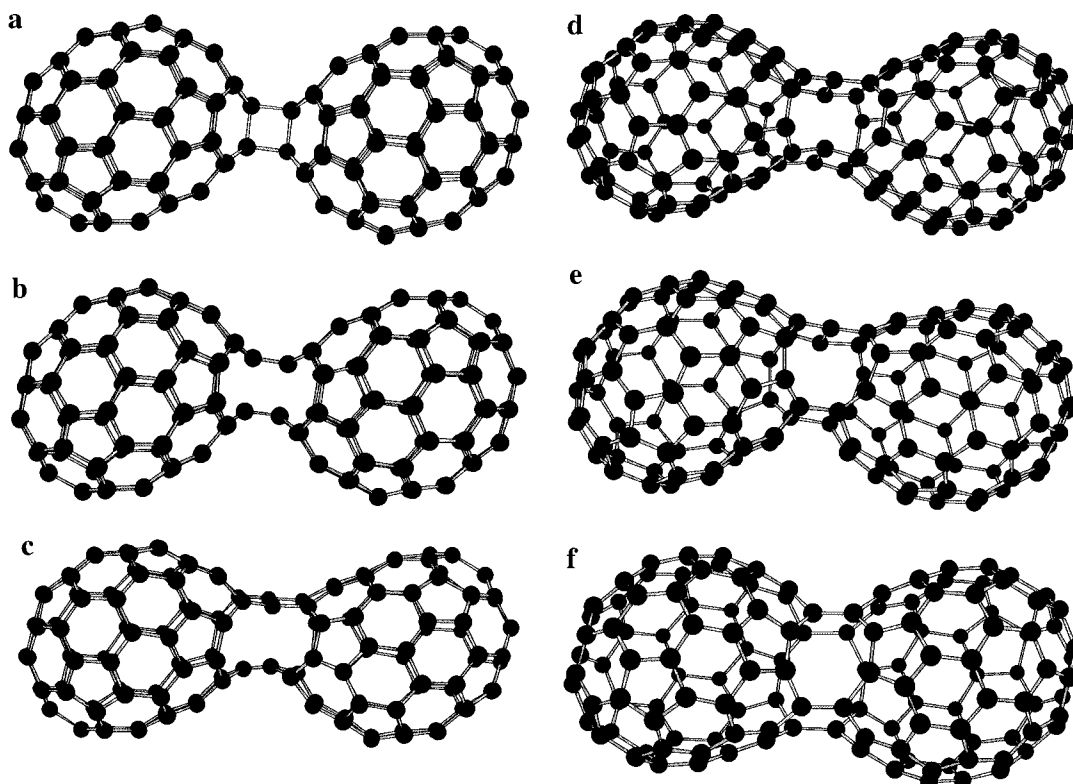


Figure 6. (a) The C_{70} dimer corresponds to $C_{140}(a)$ in Figure 4. (b) C_{140} is derived from the scission of highly strained bonds forming cyclobutane. (c) C_{140} is derived from structure b by the Stone–Wales rearrangement. (d) C_{140} is derived from structure c by further structural rearrangement of the cross-link bond. (e) C_{138} is derived from structure d by C_2 loss and recombination of dangling bonds. (f) C_{136} is derived from structure d by further C_2 loss and recombination of dangling bonds.

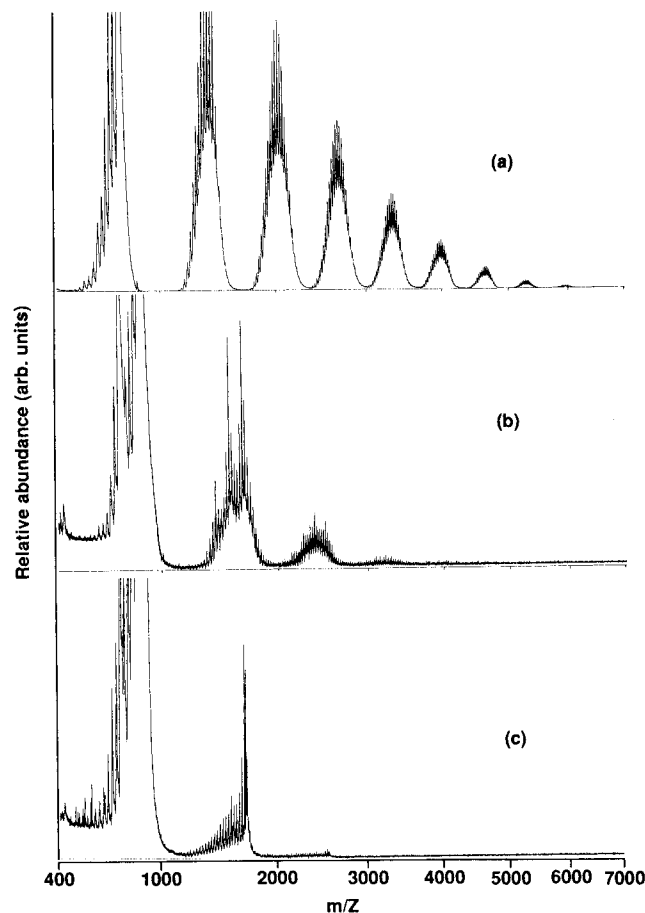


Figure 7. LDITOF-MS of C₆₀ (a, top), a 1:1 mixture of C₆₀/C₇₀ (b, middle), and C₇₀ (c, bottom) observed using a reflectron-type ion monitor.

at the same laser power the degree of polymerization of C₇₀ is obviously less than those of C₆₀ and the C₆₀/C₇₀ mixture.

The value of the electric dark current of the plasma-polymerized C₇₀ film was found to be smaller by 5 orders of magnitude than that of the C₆₀ films obtained at the same plasma power. Based on the above discussion, the insulator-like electric conductivity of the plasma-polymerized C₇₀ film can be attributed to a lower number of direct cross-links of C₇₀ molecules. The cross-link bonds, as shown in Figure 5, allow electron migration between the fullerene cages through π -conjugation, but the cross-link bonds in the sp³ valence state formed by [2+2]-cycloaddition are likely to suppress the electron migration between C₇₀ cages. Electron migration between the C₇₀ residues is impossible in an open-shell C₇₀ dimer having a cross-link bond, which has two unpaired electrons at the carbon atoms adjacent to the cross-link bond. The C₇₀ molecules which occasionally remain intact in plasma-polymerized film are responsible for the very low electric conductivity.

It follows that the direct cross-links in the plasma-polymerized C₇₀ films can be classified into two types according to their cross-link structure. One type is characterized by conductive bonds formed by C₂ loss and successive recombination resulting in π -conjugation. The other type contains insulating bonds such as found in C₇₀ dimers derived from [2+2]cycloaddition, or in open-shell C₇₀ dimer having only one cross-link bond. These two types of cross-links are interrelated in the polymer films in the sense that when the number of one type is high, the other is correspondingly low. The electric conductivity of the plasma-polymerized C₆₀ and C₇₀ films could change drastically at a certain level of formation of conducting cross-link bonds, that is, at the percolation limit.

Concluding Remarks

LDITOF mass analysis of plasma-polymerized C₇₀ films suggests that direct cross-linking of C₇₀ molecules under Ar plasma is accompanied by C₂ loss, as in the case of C₆₀.¹⁶ MO calculations of C₇₀ dimer structures suggest that a direct cross-linking occurs at the double bonds in the polyhedral caps rather than the graphitic belt of the C₇₀ molecule. The calculations also suggest that the polyhedral caps of the C₇₀ molecule are almost identical in chemical properties with the C₆₀ molecule. Furthermore, MO study of plausible C₁₃₆ molecules formed by elimination of four carbon atoms from the cyclobutane ring in the C₇₀ dimer suggests that the C₁₃₆ molecules derived from stable initial C₇₀ dimers are more stable than those derived from the unstable C₇₀ dimers.

The almost insulating electric conductivity of the plasma-polymerized C₇₀ films is attributed to a lower probability of conductive cross-linking than that of C₆₀. Taking into account the fact that the C₇₀ molecule has a graphitic belt which is less reactive than the polyhedral caps toward the addition reaction, the C₇₀ molecule can be regarded as structurally possible and as the smallest carbon nanotube, as well as a ellipsoidal fullerene with 70 carbon atoms.

Acknowledgment. The authors thank Professor Dr. E. Osawa (Toyohashi University of Technology), Dr. K. Harigaya (Electrotechnical Laboratory), Mr. P. Strasser, Dr. J. Westwater, Dr. N. Kishii, and Mr. Y. Gonno (Sony Research Center) for their fruitful discussions and help. This research is contracted with Research Institute of Innovative Technology for the Earth (RITE).

References and Notes

- (1) Kroto, H. W.; Heath, J. R.; O'Brien, S. C.; Curl, R. F.; Smalley, R. E. *Nature* **1985**, *318*, 162.
- (2) (a) Krätschmer, W.; Fostiropoulos, K.; Huffman, D. R. *Chem. Phys. Lett.* **1990**, *170*, 167. (b) Krätschmer, W.; Lamb, L. D.; Fostiropoulos, K.; Huffman, D. R. *Nature* **1990**, *347*, 354.
- (3) (a) Rao, A. M.; Zhou, P.; Wang, K.-A.; Hager, G. T.; Holden, J. M.; Wang, Y.; Lee, W.-T.; Bi, X.-X.; Eklund, P. C.; Cornett, D. S.; Duncan, M. A.; Amster, I. J. *Science* **1993**, *259*, 955. (b) Cornett, D. C.; Amster, I. J.; Duncan, M. A.; Rao, A. M.; Eklund, P. C. *J. Phys. Chem.* **1993**, *97*, 5036.
- (4) Zhou, P.; Dong, Z.-H.; Rao, A. M.; Eklund, P. C. *Chem. Phys. Lett.* **1993**, *211*, 337.
- (5) Wang, Y.; Holden, J. M.; Dong, Z.-H.; Bi, X.-X.; Eklund, P. C. *Chem. Phys. Lett.* **1993**, *211*, 341.
- (6) Ito, A.; Morikawa, T.; Takahashi, T. *Chem. Phys. Lett.* **1993**, *211*, 333.
- (7) Akselrod, L. A.; Byrne, H. J.; Thomsen, C.; Mittelbach, A.; Roth, S. *Chem. Phys. Lett.* **1993**, *212*, 384.
- (8) McElvany, S. W.; Callahan, J. H.; Ross, M. M.; Lamb, L. D.; Huffman, D. R. *Science* **1993**, *260*, 1632.
- (9) (a) Pekker, S.; Forró, L.; Mihály, L.; Jánosy, A. *Solid State Commun.* **1994**, *90*, 349. (b) Stephens, P. W.; Bortel, G.; Faigel, G.; Tegze, M.; Jánosy, A.; Pekker, S.; Oszlányi, G.; Forró, L. *Nature* **1994**, *370*, 636. (c) Chauvet, O.; Oszlányi, G.; Folló, L.; Stephens, P. W.; Tegze, M.; Faigel, G.; Jánosy, A. *Phys. Rev. Lett.* **1994**, *72*, 2721. (d) Pekker, S.; Jánosy, A.; Mihály, L.; Chauvet, O.; Carrard, M.; Forró, L. *Science* **1994**, *265*, 1077. (e) Martin, M. C.; Koller, D.; Rosenberg, A.; Kendziora, C.; Mihály, L. *Phys. Rev. B* **1995**, *51*, 3210.
- (10) Iwasa, Y.; Arima, T.; Freming, R. M.; Siegrist, T.; Zhou, O.; Haddon, R. C.; Rothberg, L. J.; Lyons, K. B.; Carter Jr., H. L.; Hebard, A. F.; Tycko, R.; Dabbagh, G.; Karajewski, J. J.; Thomas, G. A.; Yagi, T. *Science* **1994**, *264*, 1570.
- (11) (a) Muñoz-Regueiro, M.; Marques, L.; Hodeau, J. L.; Béthoux, O.; Perroux, M. *Phys. Rev. Lett.* **1995**, *74*, 278. (b) Oszlányi, G.; Folló, L. *Solid State Commun.* **1995**, *93*, 265. (c) Harigaya, K. *Phys. Rev. B* **1995**, *52*, 7968. (d) Tanaka, K.; Matsuura, Y.; Oshima, Y.; Yamabe, T. *Solid State Commun.* **1995**, *93*, 163.
- (12) Xu, C. H.; Scuseria, G. E. *Phys. Rev. Lett.* **1995**, *74*, 274.
- (13) Takahashi, N.; Dock, H.; Matsuzawa, N.; Ata, M. *J. Appl. Phys.* **1993**, *74*, 5790.
- (14) Matsuzawa, N.; Ata, M.; Dixon, D. A.; Fitzgerald, G. J. *Phys. Chem.* **1994**, *98*, 2555.
- (15) (a) Menon, M.; Subbaswamy, K. R.; Sawtarie, M. *Phys. Rev.* **1994**, *B49*, 13966. (b) Taylor, R. *J. Chem. Soc., Chem. Commun.* **1994**, 1629.

- (16) Ata, M.; Takahashi, N.; Nojima, K. *J. Phys. Chem.* **1994**, 98, 9960.
- (17) (a) Diederich, F.; Ettle, R.; Rubin, Y.; Whetten, R. L.; Beck, R.; Alvarez, M.; Anz, S.; Sensharma, D.; Wudl, F.; Khemani, K. C.; Koch, A. *Science* **1991**, 252, 548. (b) Raghavachari, K.; Rohlfing, C. M. *Chem. Phys. Lett.* **1992**, 97, 495. (c) Akasaka, T.; Mitsuhide, E.; Ando, W.; Kobayashi, K.; Nagase, S. *J. Am. Chem. Soc.* **1994**, 116, 2627.
- (18) (a) Balch, A. L.; Catalano, V. J.; Lee, J. W.; Olmstead, M. M.; Parkin, S. R. *J. Am. Chem. Soc.* **1991**, 113, 8953. (b) Hawkins, J. M.; Meyer, A.; Solow, M. A. *J. Am. Chem. Soc.* **1993**, 115, 7499.
- (19) (a) Hirsch, A.; Grösser, T.; Skiebe, A.; Soi, A. *Chem. Ber.* **1993**, 126, 1061. (b) Bingel, C. *Chem. Ber.* **1993**, 126, 1957. (c) Guldi, D. M.; Hungerbühler, H.; Wilhelm, M.; Asmus, K.-D. *J. Chem. Soc., Faraday Trans.* **1994**, 90, 1391.
- (20) (a) Borghi, R.; Lunazzi, L.; Placucci, G.; Krusic, P. J.; Dixon, D. A.; Knight, Jr. L. B. *J. Phys. Chem.* **1994**, 98, 5395. (b) Morton, J. R.; Negri, F.; Preston, K. F. *Chem. Phys. Lett.* **1994**, 218, 467. (c) Morton, J. R.; Preston, K. F.; Negri, F. *Chem. Phys. Lett.* **1994**, 221, 59. (d) Henderson, C. C.; Rohlfing, C. M.; Gillen, K. T.; Cahill, A. A. *Science* **1994**, 264, 397. (e) Cataldo, F. *Carbon* **1994**, 32, 437.
- (21) Rao, A. M.; Menon, M.; Wang, K.-A.; Eklund, P. C.; Subbaswamy, K. R.; Cornett, D. S.; Duncan, M. A.; Amster, I. J. *Chem. Phys. Lett.* **1994**, 224, 106.
- (22) (a) Hare, J. P.; Dennis, T. J. S.; Kroto, H. W.; Taylor, R.; Allaf, W. A.; Balm, S.; Walton, D. R. M. *J. Chem. Soc., Chem. Commun.* **1991**, 412. (b) Bethune, D. S.; Meijer, G.; Tang, W. C.; Rosen, H. J.; Golden, W. G.; Seki, H.; Brown, C. A.; de Vries, M. S. *Chem. Phys. Lett.* **1991**, 179, 181. (c) Dennis, T. J. S.; Hare, J. P.; Kroto, H. W.; Taylor, R.; Walton, D. R. M.; Hendra, P. J. *Spectrochim. Acta* **1991**, 47A, 1289. (d) Christides, C.; Nikolaev, A. V.; Dennis, T. J. S.; Prassides, K.; Negri, F.; Orlandi, G.; Zerbetto, F. *J. Phys. Chem.* **1993**, 97, 3641. (e) Jishi, R. A.; Dresselhaus, M. S.; Dresselhaus, G.; Wang, K.-A.; Zhou, P.; Rao, A. M.; Eklund, P. C. *Chem. Phys. Lett.* **1993**, 206, 187. (f) van Loosdrecht, P. H.; Verheijen, M. A.; Meekes, H.; van Bentum, P. J. M.; Meijer, G. *Phys. Rev. B* **1993**, 47, 7610.
- (23) Yoshikawa, M.; Nagai, N.; Matsuki, M.; Fukuda, H.; Katagiri, G.; Ishida, I.; Isitani, A.; Nagai, I. *Phys. Rev. B* **1992**, 46, 7169.
- (24) Stewart, J. J. P. *J. Comput.-Aided Mol. Des.* **1990**, 4, 1.
- (25) Dewar, M. J. S.; Zoebisch, E. G.; Healy, E. F.; Stewart, J. J. P. *J. Am. Chem. Soc.* **1985**, 107, 3902.
- (26) Stewart, J. J. P. QCPE Program 455, 1983; Version 5.10.
- (27) (a) Wilson, S. R.; Kaprinidis, N.; Wu, Y.; Schuster, D. I. *J. Am. Chem. Soc.* **1993**, 115, 8495. (b) Zhang, X.; Romero, A.; Foote, S. J. *J. Am. Chem. Soc.* **1993**, 115, 11024.
- (28) Locally developed software was kindly provided by Dr. N. Kishii of the Sony Corp. Res. Center.
- (29) (a) Murry, R. L.; Strout, D. L.; W. C.; Odom, G. K.; Scuseria, G. E. *Nature* **1993**, 366, 665. (b) Strout, D. L.; Murry, R. L.; Xu, C.; Eckhoff, W. C.; Odom, G. K.; Scuseria, G. E. *Chem. Phys. Lett.* **1993**, 214, 576. (c) Eckhoff, W. C.; Scuseria, G. E. *Chem. Phys. Lett.* **1993**, 216, 399.
- (30) Osawa, E., private communication.
- (31) (a) Stone, A. J.; Wales, D. J. *Chem. Phys. Lett.* **1986**, 128, 501. (b) Saito, R. *Chem. Phys. Lett.* **1992**, 195, 537. (c) Saito, R.; Dresselhaus, G. D.; Dresselhaus, M. S. *Chem. Phys. Lett.* **1992**, 195, 537. (d) Osawa, S.; Osawa, E. *Fullerene Sci. Technol.* **1995**, 3, 565.
- (32) Fieber-Erdmann, M.; Krätschmer, W.; Ding, A. Z. *Phys. D* **1993**, 26, 308.
- (33) Petrie, S.; Bohme, D. K. *Nature* **1993**, 356, 426.
- (34) (a) Krusic, P. J.; Wasserman, E.; Parkinson, B. A.; Malone, B.; Holler, E. R.; Keizer, P. N.; Morton, J. R.; Preston, K. F. *J. Am. Chem. Soc.* **1991**, 113, 6274. (b) Morton, J. R.; Preston, K. F.; Krusic, P. J.; Hill, S. A.; Wasserman, E. *J. Phys. Chem.* **1992**, 96, 3576. (c) Keizer, P. N.; Morton, J. R.; Preston, K. F. *J. Chem. Soc., Chem. Commun.* **1992**, 1259.
- (35) (a) Krusic, P. J.; Wasserman, E.; Keizer, P. N.; Morton, J. R.; Preston, K. F. *Science* **1991**, 254, 1183. (b) Morton, J. R.; Preston, K. F.; Krusic, P. J.; Hill, S. A.; Wasserman, E. *J. Am. Chem. Soc.* **1992**, 114, 5454.
- (36) McEwen, C. N.; McKay, R. G.; Larsen, B. S. *J. Am. Chem. Soc.* **1992**, 114, 4412.