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(54) **REACTORS FOR PRODUCING INORGANIC FULLERENE-LIKE TUNGSTEN DISULFIDE HOLLOW NANOPARTICLES AND NANOTUBES**

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REACTEURS POUR LA PRODUCTION DE NANOPARTICULES ET NANOTUBES CREUX A BASE  
DE DISULFURE DE TUNGSTENE DE TYPE FULLERENE INORGANIQUE

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## Description

### FIELD OF THE INVENTION

**[0001]** This invention is generally in the field of production of inorganic fullerene-like nanoparticles (*IF*) and nanotubes. More particularly it relates to reactors for production of tungsten disulfide hollow onion-like nanoparticles and nanotubes.

### BACKGROUND OF THE INVENTION

**[0002]** Inorganic fullerene-like (*IF*) nanoparticles and nanotubes are attractive due to their unique crystallographic morphology and their interesting physical properties. In particular, disulfides of molybdenum and tungsten belong to a class of solid lubricants useful in vacuum, space and other applications where liquids are impractical to use. The fullerene-like nanoparticles can be used as superior solid lubricants, e.g. as additives to fluids and for self-lubrication.

**[0003]**  $MS_2$  ( $M=Mo, W$ ) layered compounds can be synthesized by sulfidization of the respective trioxides in a reducing atmosphere at elevated temperatures. Hollow  $MoS_2$  and  $WS_2$  onion-like (inorganic fullerene-like, *IF*) nanoparticles were first observed in thin ( $\sim 20$ nm) films, which were formed by sulfidization of the respective amorphous  $MO_3$  films in a reducing atmosphere at temperatures of about  $850^\circ C$  (Tenne, R., Margulis, L., Genut, M., and Hodes, G., *Nature* **360**, 444 (1992); Margulis, L., Salitra, G., Tenne, R., and Talianker, M., *Nature* **365**, 113 (1993)). In a search for the synthesis of a pure *IF* phase, it was suggested to take oxide powder rather than a thin film as a precursor material. However,  $MoO_3$  powder evaporates at temperatures above  $700^\circ C$ , while  $WO_3$  does not sublime until  $1400^\circ C$ . Therefore, at the relevant reaction temperatures ( $\sim 850^\circ C$ ), the reduction/sulfidization reactions of  $MoO_3$  and  $WO_3$  powders occur by the gas-phase reactions (GPR) and solid-gas reactions (SGR), respectively.

**[0004]** In the case of the GPR, the size and shape of the final reaction product depends only on the prevalent conditions in the reactor, since  $MoO_3$  evaporates as  $(MoO_3)_{3-5}$  molecular clusters (Magneli, A., *J. Inorg. Nucl. Chem.* **2**, 330 (1956)). The reactor, which was used in an earlier report by some of the inventors of the present invention, (Feldman, Y., Wasserman, E., Srolovitz, D.J., and Tenne, R., *Science* **267**, 222 (1995)) was very simple, indeed. A mixture of the three gases ( $N_2$ ,  $H_2$ ,  $H_2S$ ) was made to flow through the  $MoO_3$  vapor zone. *IF*- $MoS_2$  powder accrued on the reactor walls along with a large number of other oxide and sulfide phases obtained by the GPR, when the  $MoO_3$  powder with crystallites of  $\sim 5\mu m$  were used as a precursor. To regulate the GPR process and increase the production yield of *IF*- $MoS_2$ , it was decided to separate the sublimation of  $MoO_3$  powder from its sulfidization reaction. Several modifications of the reactor enabled the preparation a few milligrams of

an almost pure *IF* phase in a single run (Feldman, Y., Wasserman, E., Srolovitz, D.J., and Tenne, R., *Science* **267**, 222 (1995)). By varying the annealing time, the intermediate products of the reaction could be identified.

**[0005]** It should be understood that, according to the known mechanism for the growth of the *IF* phase in molybdenum or tungsten systems, the analysis of the kinetics of the simultaneous reduction and sulfidization of  $WO_3$  powders are based on the occurrence of a unique driving force for the fast growth of the first curved sulfide layer (0001) around an oxide nanoparticle. According to that kinetic model, a synergy between the reduction and sulfidization processes, which occurs in a very narrow window of parameters, leads to the formation of the first closed sulfide layers (Feldman, Y., Lyakhovitskaya, V., and Tenne, R., *J. Am. Chem. Soc.* **120**, 4176 (1998)). On the basis of this model, the inventors have estimated the temperature regimes for the formation of small (10-30nm) or large (100-200nm) *IF* of  $WS_2$ ,  $WSe_2$ , and  $WTe_2$  materials, and experimental results have shown a good agreement with the predictions of the "synergy" model [Feldman, Y., Lyakhovitskaya, V., and Tenne, R., *J. Am. Chem. Soc.* **120**, 4176 (1998); Tenne, R., Homyonfer, M., and Feldman, Y., *Adv. in Metal and Semiconductor clusters*, Ed. M.A. Duncan, JAI Press Inc. **4**, 227 (1997)]. The general principles of the sulfidization of the respective oxides, of the growth model were successfully applied also to the growth of *IF* in other layered systems, like  $VS_2$ ,  $In_2S_3$ , and  $SnS_2$  [Tsirlina, T., Feldman, Y., Homyonfer, M., Sloan, J., Hutchison, J.L., and Tenne, R., *Fullerene Science & Technology* **6**, 157 (1998); Homyonfer, M., Alpers, B., Rosenberg, Yu., Sapir, L., Cohen, S.R., Hodes, G., and Tenne, R., *J. Am. Chem. Soc.* **119**, 2693 (1997)].

**[0006]** It is necessary to point out here that the method of *IF* (including the nanotubes) formation is a "chemical" one: i.e., a chemical reaction is essential for the growth of these nanoparticles. Following this early work by the inventors of the present invention, a few reports on the synthesis of  $MoS_2$  nano and microtubes by other "chemical" methods have appeared in the literature, recently [Remskar, M., Skraba, Z., Cleton, F., Sanjines, R., and Levy, F., *Surf. Rev. Lett.* **5**, 423 (1998); Remskar, M., Skraba, Z., Regula, M., Ballif, C., Sanjines, R., and Levy, F., *Adv. Mat.* **10**, 246 (1998); Vollath, D., and Szabo, D.V., *Mater. Lett.* **35**, 236 (1998); Zelenski, M., and Dorhout, P.K., *J. Am. Chem. Soc.* **120**, 734 (1998)].

**[0007]** It should be emphasized that the synthesis of carbon fullerenes using "physical" methods, implies curvature of very small atomic sheets followed by the annihilation of the dangling bonds of the peripheral atoms, a process, which is induced by a high rate of energy dissipation. It was shown recently that *IF*- $MoS_2$  could be obtained also by "physical" methods, such as e-beam irradiation (Jose-Yacamán, M., Lorez, H., Santiago, P., Galvan, D.H., Garzon, I.L., and Reyes, A., *Appl. Phys. Lett.* **69**, 8, 1065 (1996), or laser ablation of regular  $MoS_2$  powder (Parilla, P.A., Dillon, A.C., Jones, K.M., Riker, G.,

Schulz, D.L., Ginley, D.S., and Heben, M.J., *Nature* **397**, 114 (1999), or by short electrical pulses from the tip of a scanning tunneling microscope through amorphous MoS<sub>3</sub> nanoparticles (Homyonfer, M., Mastai, Y., Hershfinkel, M., Volterra, V., Hutchison, J.L., and Tenne, R., *J. Am. Chem. Soc.* **118**, 33, 7804 (1996). *IF*-NiCl<sub>2</sub>, including nanotubes were recently observed after heating of NiCl<sub>2</sub> at 960°C in a reducing atmosphere (Rosenfeld-Hacohen, Y., Grunbaum, E., Sloan, J., Hutchison, J.L., and Tenne, R., *Nature* **395**, 336 (1998). However, the *IF* yield in these cases is very modest and can not be compared with the large amounts of the pure *IF*-WS<sub>2</sub> (MoS<sub>2</sub>) phase obtained by "chemical" methods.

**[0008]** The first SGR reactor for the synthesis of macroscopic quantities of *IF*-WS<sub>2</sub> was constructed in 1996 based on the principles of the above indicated reaction mechanism. To increase the amount-of the reactant (oxide) and expose its entire surface to the gas, a bundle of quartz tubes was placed inside the main quartz tube (40mm diameter) and the oxide powder was dispersed in them, very loosely. The reactor was introduced into a horizontal furnace and the powder was placed in a constant temperature region (~850°C). A stream of H<sub>2</sub>/N<sub>2</sub> + H<sub>2</sub>S gases passed through every tube filled with the powder. It created rather equivalent conditions for the reaction of the entire oxide powder during SGR. Typically, about 0.4 gram of *IF*-WS<sub>2</sub> could be obtained in a single batch (3 hr), with a conversion yield of almost 100%.

**[0009]** The precursor (oxide) powder and the reaction products were analyzed by X-ray powder diffraction (XRD) and transmission electron microscopy (TEM). The inventors recall that the size and shape of the precursor tungsten oxide nanoparticles predetermine the *IF*-WS<sub>2</sub> dimensions in the SGR synthesis. The starting material for the synthesis of *IF*-WS<sub>2</sub> was a WO<sub>3</sub> powder (>99% pure), with almost spherical particles having sizes smaller than ca. 0.2μm. Oxide powder having a larger particle size was converted mainly into 2H-WS<sub>2</sub> phase. Moreover, 2H-WS<sub>2</sub> platelets were predominantly obtained if the packing of the powder was too compact. This fact required a very thorough control of the *IF* quality after every batch. WO97/44278 reports a method and an apparatus for preparing WS<sub>2</sub> fullerene-like molecules in which WO<sub>3</sub> particles are deposited on the bottom of a reactor and are dispersed by a flow of N<sub>2</sub>+H<sub>2</sub> which blows from behind. The reduced particles react then with H<sub>2</sub>S.

**[0010]** It was hypothesized before, that the tribological properties of the *IF* nanoparticles are closely related to their structure [(Rapoport, L., Bilik, Yu., Feldman, Y., Homyonfer, M., Cohen, S.R., and Tenne, R., *Nature* **387**, 791 (1997). See also, *Nature* **387**, 761 (1997); and Rapoport, L., Feldman, Y., Homyonfer, M., Cohen, H., Sloan, J., Hutchison, J.L., and Tenne, R., *Wear* **225-229**, 975 (1999)]. In general, spherical *IF* nanoparticles exhibited tribological properties superior to their 2H counterpart, while their performance rapidly deteriorated upon loss of spherical shape. Therefore, it is expected that the more spherical *IF*-WS<sub>2</sub> nanoparticles would perform very well

as solid lubricants. In addition, the larger *IF* should be a better lubricant in the case of the friction of two matting metal surfaces having higher surface roughness.

## 5 SUMMARY OF THE INVENTION

**[0011]** There is accordingly a need in the art to facilitate and optimize the solid-gas chemical reactions, which would increase the amount of available inorganic fullerene-like nanostructures.

**[0012]** The present invention thus provides a new reactor, which allows for a scale-up of *IF*-WS<sub>2</sub> production including nested fullerene-like structures and WS<sub>2</sub> nanotubes by more than two orders of magnitude. The falling-bed reactor, and especially fluidized-bed-designed reactors, pave the way for an almost ideal growth conditions for synthesizing *IF* from an oxide precursor. As a result, the produced *IF* nanoparticles have a more uniform spherical shape and can grow to a larger size (up to 0.5μm). Likewise, the produced of nanotubes have a uniform size distribution. It is expected that the relatively spherical *IF*-WS<sub>2</sub> nanoparticles, which are produced by the falling or the fluidized bed reactor, will exhibit superior tribological properties, than reported before.

**[0013]** Thus, according to one aspect of the present invention, there is provided an apparatus for the production of inorganic fullerene-like (*IF*) nanoparticles and nanotubes, the apparatus comprising a chemical reactor, and is associated with a feeding set up and with a temperature control means for controlling the temperature along the reaction path inside the reactor so as to maintain the temperature to be substantially constant, the chemical reactor comprising:

- (i) a flow line for providing a continuous flow of a mixture of a reacting metal oxide powder and an inert flow assisting gas from the feeding setup into the reaction path, which extends along the reaction path;
- (ii) at least one inlet port for supplying reacting gases into the reaction path to flow therethrough along the reaction path, an interaction between the reacting gases and said mixture resulting in the formation of the *IF* nanostructures and nanotubes during the flow of the interacting materials along the reaction path;
- (iii) at least one outlet port for trapping reacting gases; and
- (iv) at least one collector located at a the opposite end portion of the reactor for collecting the formed *IF* nanostructures or nanotubes.

The feeding set-up comprises a flexible portion by which it is connected to the flow line of the reactor. The feeding set-up comprises two flexibly inter-connected containers aligned along an axis of feeding the mixture, each container having a mesh at its bottom and being mounted on a vibration assembly, the lower one of the containers being connected to the flow line of the reactor. The reaction path may extend along a vertical axis of the reactor

and thus the at least one collector is located at the bottom of the reaction path. The inlet port for supplying the reacting gases which their flow is assisted by inert gas is located upstream of the reaction path. In a preferred embodiment this inlet for supplying the reacting gases is located downstream of the reaction path, and the flow of the reacting gases towards and through the reaction path being assisted by an inert gas. The temperature along the reaction path should be constant and may vary from about 750°C to about 850°C. The resulting nanoparticles may be *IF*-WS<sub>2</sub> nanoparticles or a combination of *IF*-WS<sub>2</sub> nanoparticles and WS<sub>2</sub> nanotubes.

**[0014]** According to yet another aspect of the invention there is provided a synthetic method of production of WS<sub>2</sub> fullerene-like nanoparticles and nanotubes said method comprising the steps of:

- (i) providing a chemical reactor defining a reaction path with predetermined temperature conditions along the reaction path;
- (ii) feeding into the chemical reactor a mixture of a reacting tungsten oxide powder and an inert gas with a predetermined feeding rate, and feeding reacting gases so as to provide a continuous flow of the mixture and the gases through the reaction path;
- (iii) maintaining the predetermined temperature to be substantially constant, thereby providing a solid-gas reaction between the reacting gases and the tungsten oxide powder during the flow through the reaction path, resulting in the formation of WO<sub>3-x</sub> clusters coated by at least one layer of WS<sub>2</sub>, and further exchange of oxides by sulfides in inner parts of the clusters, thereby causing formation of WS<sub>2</sub> nanostructures; and
- (iv) collecting the formed WS<sub>2</sub> nanostructures.

**[0015]** The feeding rate of the reacting tungsten oxide powder is selected in accordance with the dimensions of the chemical reactor, thereby obtaining the WS<sub>2</sub> nanostructures in the form of nanoparticles, or a combination of nanoparticles and nanotubes. The feeding rate ranges is between about 3mg/min to about 120mg/min, the reacting gases being H<sub>2</sub> and H<sub>2</sub>S, the concentration of the H<sub>2</sub> gas being from about 0.2 % (w/v) to about 7 % (w/v) and the concentration of the H<sub>2</sub>S gas being from about 0.1 % (w/v) to about 6 % (w/v). By predetermining the appropriate feed rate and appropriate concentration, either *IF*-WS<sub>2</sub> nanoparticles may exclusively be obtained or a combination of WS<sub>2</sub> nanotubes and *IF*-WS<sub>2</sub> nanoparticles may be obtained. The resulting nanoparticles, each have a substantially perfect spherical shape with a size of up to 0.5µm. The resulting nanotubes which may be obtained each having a length of up to a few hundreds µm and a cross-sectional dimension of up to 200 nanometer.

## BRIEF DESCRIPTION OF THE DRAWINGS

**[0016]** In order to understand the invention and to see how it may be carried out in practice, a preferred embodiment will now be described, by way of non-limiting example only, with reference to the accompanying drawings, in which:

**Fig. 1** is a two-stage vibro-electromagnet feeding set-up suitable for use in an apparatus according to the invention;

**Figs. 2A and 2B** illustrate, respectively, a schematic representation of a falling bed reactor including the set-up of Fig. 1, and the temperature profile along the vertical axis of the reactor;

**Fig. 2C** illustrates a typical TEM image of W<sub>5</sub>O<sub>14</sub> whiskers produced in the reactor according to the invention and serving as precursors for the production of WS<sub>2</sub> nanotubes;

**Figs. 3A and 3B** illustrate, respectively a schematic representation of a fluidized bed reactor including the set-up of Fig. 1, and the temperature profile along the vertical axis thereof;

**Figs. 4A and 4B** illustrate a TEM image of the typical *IF*-WS<sub>2</sub> nanoparticles, and high magnification of two *IF*-WS<sub>2</sub> nanoparticles;

**Fig. 5** is a SEM image of an ensemble of *IF*-WS<sub>2</sub> nanoparticles; and

**Fig. 6** illustrates a typical TEM image of *IF*-WS<sub>2</sub> nanotubes.

## DETAILED DESCRIPTION OF A PREFERRED EMBODIMENT

**[0017]** The present invention deals with a method for the optimized production of inorganic fullerene-like nanoparticles and nanotubes. The synthetic route is a solid-gas reaction and two reactors (falling bed reactor and fluidized bed reactor) are utilized for the production of either the nanoparticles or the nanotubes.

**[0018]** The chemical reaction that occurs in the reactors of the present invention leading to the formation of the nanoparticles and nanotubes is the following. The metal oxide, WO<sub>3</sub>, nanoparticles as agglomerates are derived downwards from a feeding set-up with the aid of a N<sub>2</sub> stream. The descending WO<sub>3</sub> agglomerates interact with a mixture of H<sub>2</sub> and H<sub>2</sub>S gases. The H<sub>2</sub>, being lighter, diffuses faster than the H<sub>2</sub>S and interacts initially with the descending WO<sub>3</sub> partially reducing the agglomerates. Concurrently, the WO<sub>3</sub> interacts with the H<sub>2</sub>S stream thus the overall interaction of the WO<sub>3</sub> agglomerates with the H<sub>2</sub> and H<sub>2</sub>S gases results in the formation of a completely closed monomolecular WS<sub>2</sub> layer around each nanoparticles. The formation of the first closed atomic layer of WS<sub>2</sub> on the surface of the metal oxide nanoparticle occurs during the short time interval between the beginning of the descending of the WO<sub>3</sub> agglomerates onto the top of the reactor (which process is N<sub>2</sub> assisted) and until

the so-produced WS<sub>2</sub> coated WO<sub>3-x</sub> nanoparticles land on the filter (in the case of the falling bed reactor). This primary step is essential for the success of the IF synthesis, since it passivates the nanoparticles surface and prohibits coalescence of the nanoparticles, once they land on the bottom of the reactor. In the case of a fluidized bed reactor, no agglomeration of the oxide clusters could occur, since the nanoparticles are fluidized in the gas atmosphere during the entire reaction time.

**[0019]** During the creation of a completely closed monomolecular WS<sub>2</sub> layer (several seconds), a rather fast reduction of the oxide core by hydrogen occurs together with a fast deintercalation of oxygen and formation of crystal shears, i.e. rows of edge sharing octahedra. In the next step a slow intercalation of sulfur and its diffusion to the growth front occurs, the WO<sub>3</sub> being thus converted into tungsten sulfide. The latter process may last 30-300min depending on the nanoparticle's size and the reaction temperature. In the case of nanoparticles smaller than 100nm, this later process may last 60-120 min. The use of relatively large oxide particles (about 0.5μm), in which sulfur intercalation is rather slow, necessitates an addition of an extra annealing step at 950°C (5 hr) at the end of the present process, in order to complete the conversion of the oxide core into tungsten sulfide.

**[0020]** Consequently, the size of the IF- WS<sub>2</sub> particle is determined by the size of the incipient WO<sub>3</sub> nanoparticle. Furthermore, the geometry of the resulting IF-WS<sub>2</sub> nanoparticles is determined by the geometry of the WO<sub>3</sub> particles undergoing the reduction, initial interaction with H<sub>2</sub>S, and further diffusion of sulfur atoms replacing the oxygen atoms. It is important to note that each IF-WS<sub>2</sub> nanoparticle has a single growth front of the sulfide, and therefore each nanoparticle is a single crystallite.

**[0021]** It is thus apparent that the mechanism of formation of the WS<sub>2</sub> nanotubes is slightly different, since the initial geometry of the reacting WO<sub>3</sub> particle should already reflect its final structure. In the formation of the IF-WS<sub>2</sub> nanotubes, the initial WO<sub>3</sub> particles interact under the terms that exist in the reactor with the slight humidity present, which is produced *in-situ* by the reduction of the metal oxide nanoparticles with H<sub>2</sub> present, leading to the formation of WO<sub>3-x</sub>(H<sub>2</sub>O)<sub>y</sub> particles. These particles sublimize and condense to form W<sub>5</sub>O<sub>14</sub> nanoparticles which are in the form of whiskers (needles). These formed whiskers then undergo the partial reduction with H<sub>2</sub> gas and the further interaction with the H<sub>2</sub>S gas to form a core of W<sub>18</sub>O<sub>49</sub> covered completely with a first layer of sulfide (A. Rothschild, J. Sloan, and R. Tenne, J. Am. Chem. Soc. 122, 5169 (2000)). The next step is exactly as the one happening in the formation of IF-WS<sub>2</sub> nanoparticles, i.e. a slow diffusion-controlled reaction of converting the inner oxide into sulfide layers.

**[0022]** It should be understood that the growth mechanism of IF- WS<sub>2</sub> was elucidated in quite a detail manner [the above-mentioned articles: Feldman, Y, Frey, G.L., Homyonfer, M., Lyakhovitskaya, V., Margulis, L., Cohen, H., Hodes, G., Hutchison, J.L., and Tenne, R., J. Am.

Chem. Soc. **118**, 5362 (1996); Feldman, Y, Lyakhovitskaya, V, and Tenne, R., J. Am. Chem. Soc. **120**, 4176 (1998)], and therefore should not be described here in details. It was found that, under certain conditions, the simultaneous reduction and sulfidization of oxide nanoparticles lead to IF-WS<sub>2</sub> formation.

**[0023]** However, it appears that, if some process parameters are changed and either of the above two processes prevails, no fullerene-like particles will be synthesized. In particular, a deviation of the hydrogen/sulfur concentration ratio from the specified value at any point of the reaction chamber will most likely lead to 2H-WS<sub>2</sub> platelets formation there. Therefore, it is very important, especially at the first instant of the reaction, that a homogeneous mixture of the three gases would engulf each oxide nanoparticle. The falling bed and fluidized reactors of the present invention are most suited and are optimized such as to ensure that all required conditions, are achieved and are maintained throughout the reaction.

**[0024]** Fig. 1 illustrates a feeding set-up **10** suitable for use in the apparatus of the present invention. The set-up **10** is a vibro-electromagnet set-up constructed for co-feeding the WO<sub>3</sub> powder into a quartz tube based reactor in two stages, in a predetermined rate. The set-up consists of two glass beakers **12** and **14** with a mesh at their bottom, **12a** and **14a**, respectively, which are placed one on top of the other. The upper beaker **12** provides a coarse portion of WO<sub>3</sub> powder to the lower beaker **14**, which then feeds the oxide powder into the reactor in fine steps. For that to occur, the bottom of the upper beaker **12** was joined to the entrance of the lower glass beaker **14** by a flexible rubber balloon **16**. A similar flexible balloon connects the bottom of the lower glass beaker **14** to the top entrance of the reactor (not shown here). The soft balloons **16** serve also to dump the mechanical vibrations between the two beakers, and between them and the reactor. Each glass beaker is mounted on an electromagnet **18a** and **18b** (constituting a vibrating assembly), which shakes the respective glass beaker in a fully controllable rate. As shown, nitrogen gas is flushed through the feeding glass beakers continuously from up side down and assists the lowering of the oxide powder.

**[0025]** The shaking of the two glass beakers lead to dropping of a weighted amount of the WO<sub>3</sub> powder through the meshes at the bottom of each beaker, **12a** and **14a**, respectively. The rate of shaking of the higher electromagnet is inferior to that of the lower one. However, since the higher beaker **12** has a larger mesh-size **12a** in its bottom, it can deliver, as a result of a single shake, a relatively large amount of material to the lower beaker **14**. The lower beaker, in its turn, is able to deliver a much smaller portion of material at each shake, due to the fine mesh size in its bottom **14a**. Typically, the rate of shaking of the upper beaker **12** is 1 pulse/min, while the rate of shaking of the lower beaker **14** is 15 pulse/min. However, while the upper beaker **12** may deliver 30mg per pulse, the lower beaker **14** may deliver only 2mg per pulse.

**[0026]** This vibro-electromagnet set-up **10** secures a continuous flux of even very small amount of metal oxide powder (from about 0.2 g/hr) to the reactor. The setup enables the metal oxide powder to go from the upper glass beaker **12** to the reactor without contacting the atmosphere. The feeding rate of the oxide powder is computer controlled and could be varied in the range from about 3 mg/min to about 120mg/min. Careful selection of the feeding parameters (mesh-size, vibration rate and amplitude, rate of gas flow, etc.), are mandatory in order to avoid the accumulation of the metal oxide powder in one of the stages and provide the reactor with a continuous yet small flux of metal oxide precursor.

**[0027]** Fig. 2A illustrates a falling bed apparatus **20**, according to the invention, for synthesizing the *IF* nanoparticles and nanotubes. The apparatus **20** is in the form of a quartz tube reactor (40mm diameter), which is placed inside a vertical furnace **21** (constituting a temperature control means). The furnace is designed and operated so as to define two heating zones for the purposes of the present invention. It should, however, be noted, that the present invention is not limited to the use of two heating zones, and this particular furnace is constructed with two heating zones in order to maintain the temperature inside the reactor constant. Any setting that would keep the temperature constant is possible.

**[0028]** Placed at the top of the apparatus (reactor) **20** is the vibro- electromagnet **10** set-up for cofeeding the  $WO_3$  powder with the aid of  $N_2$  into the reactor. The reactor **20** is formed with an annular inlet opening **22** for supplying a mixture of reacting gases,  $H_2S$  and  $H_2$ , the flow of which towards the inside of the reactor is driven by  $N_2$ . A tube (flow line) **23** connects the feeding set-up **10** to the inside of the reactor **20**, thereby providing a precursor of the metal oxide powder to the reactor. A filter (collector) **24** is mounted at the bottom portion of the reactor **20** for collecting the end product. The escaping gases are trapped through an outlet opening **26**.

**[0029]** The cofeeder delivers the  $WO_3$  powder from the top into the vertical reactor **20**, and the mixture ( $WO_3$  powder and  $N_2$ ) flows in the same direction. This technique enables to minimize the possibility of any physical contact between the oxide nanoparticles at the early stages of the reaction. Moreover, this method rendered the synthetic process quasi-continuous. At the top of the reactor, the dropping metal oxide powder intersects with a flow of  $H_2S$  and  $N_2/H_2$  gasses entering the reactor at the opening **22**. The interaction leads to the formation of the nanoparticles of  $WO_{3-x}$  covered with a first  $WS_2$  layer, which are collected on the filter **24**. In order to obtain *IF*- $WS_2$  nanoparticles the feed of  $WO_3$  should be in the range of from about 3mg/min to about 40mg/min. The total stream rate of the gasses is in the range of about 150cc/min. The concentration of the  $H_2S$  and  $H_2$  stream is in the range of from about 3 % to about 7 % (v/w). The temperature should be kept constant throughout the entire reaction path (variation of  $\pm 5^\circ C$ ). The temperature in the reaction path may however be in the range from

about  $750^\circ C$  to about  $850^\circ C$  depending on the particle size of the metal oxide ( $WO_3$ ) precursor. The pressure in the reactor is in the range of about 1.1 atm. The pressure is determined by the traps located next to the exit **26**.

**[0030]** The temperature profile along the reaction path (z-axis) inside the reactor is illustrated in Fig. 2B. It is evident that the temperature during the various process stages is maintained constant. It should be noted that precise control of the reaction parameters is indispensable for the complete conversion of the tungsten oxide powder into *IF* tungsten sulfide nanoparticles. Radial and axial deviations of the temperature along the reaction path do not exceed  $\pm 5$  degrees. Such variations in temperature are allowable by *IF*- $WS_2$  synthesis.

**[0031]** Under the above experimental parameters,  $WS_2$  nanotubes be obtained as a mixture together with *IF*- $WS_2$  nanoparticles in case the feed of  $WO_3$  is increased by 3 to 5, i.e. in the range of from about 40mg/min to about 120mg/min. On the other hand, such a mixture of nanotubes together with *IF*- $WS_2$  nanoparticles may be obtained with a feed rate in the range of from about 3mg/min to about 40mg/min. However, in such a case the other parameters have to be altered. Thus, in order to obtain  $WS_2$  nanotubes, the concentration of the  $H_2$  stream should be in the range of from about 0.2 % to about 4 % (v/w). The  $H_2S$  concentration should be in the range of from about 0.2 % to about 4 %. The change in the experimental parameters leading to the formation of nanotubes affects the clusters undergoing the reduction and the interaction with  $H_2S$ .

**[0032]** As a result of this effect,  $W_5O_{14}$  nanoparticles (or other  $WO_{3-x}$  phases) are created being in the form of whiskers. Fig. 2C shows a TEM image of the  $W_5O_{14}$  whiskers which serve as the precursor powder, formed *in-situ*, for obtaining the nanotubes. As indicated above, although not specifically shown, these formed whiskers then undergo the partial reduction with  $H_2$  gas and an interaction with the  $H_2S$  gas to form a core of  $W_{18}O_{49}$  covered completely with a first layer of sulfide.

**[0033]** Thus, the resulting powder, nanoparticles or nanotubes are collected on the filter **24**, which is placed at the end of the constant temperature path ( $830^\circ C$  according to the present experimental parameters), and further react with  $H_2S$  in order to obtain fully sulfidated nanostructures.

**[0034]** Obviously, since the process is quasi-continuous, the amount of *IF* increases with time during the synthesis. However, when the amount of *IF* per batch exceeds 15 grams, the quality of *IF* nanoparticles deteriorates. It was found by the inventors that, if the thickness of the powder on the filter **24** is too large, rather big clotted lumps start to appear, and, as a result,  $2H-WS_2$  platelets are formed. This can be overcome by the use of a fluidized bed reactor based apparatus.

**[0035]** Fig. 3A illustrates such a fluidized bed apparatus **30** for the *IF* synthesis utilizing the feeding set-up **10** for feeding the oxide powder towards the inside of the apparatus. The apparatus **30** is a quartz tube reactor

installed in a furnace **31**. Here, three heating zones are used aligned along the vertical axis of the reactor. The use of three heating zones is by no means limiting, and, generally, any setting that would keep the temperature constant along the reaction path is possible. A long quartz tube **32** and a middle tube **36** are provided. The tube **32** connects the feeding set-up **10** to the interior of the reactor inside the tube **36**, and extends through the upper heating zone and down to the reaction zone inside the tube **36**. As shown, the tube **36** is conical-shaped having its wide part **38**. A stream of reacting gases,  $H_2$  and  $H_2S$ , is provided through an inlet opening **34** formed at the bottom of the reactor. Further provided in the reactor **30** is a filter **40** placed inside the tube **36** at the end of the constant temperature region ( $830^\circ C$  according to the present experimental conditions), and an additional filter **42** outside the tube **36**, both for collecting a small part of powder propagating towards the filters. Escaping gases are discharged through an outlet opening **44**.

**[0036]** In the present example, the quartz tube has a length of 60cm and a diameter of 50mm. The flow of the oxide powder  $WO_3$ , assisted by nitrogen gas flow (about 60cc/min), passes through the 12mm diameter and quartz tube **32**, which may be from about 15cm to about 30cm long (30cm according to the present reactor) within the reactor. The stream of the reacting gases,  $H_2$  and  $H_2S$  enters the reactor from the bottom opening **34** and flows upwards in a conical-shaped inner quartz tube through the middle part **36** having a diameter of about 22mm towards its wider portion **38**. The rate of the stream of the gas  $H_2S$  and  $N_2/H_2$  is from about 100cc/min to about 200cc/min. Therefore, the linear gas flow rate in the wide part **38** of the tube is approximately from about 10cm/min to about 30cm/min, which is about ten times slower than that in the narrow part of the tube **36**. The flow of the mixture of  $N_2$  and the oxide powder  $WO_3$  leaps out the tube **32**, and interfuses with the flow of  $H_2S$  and  $N_2/H_2$  gases in the wide part **38** of the conical-shaped tube. The reduction and sulfidization of the  $WO_3$  particles starts as a result of the interfusion. The interfusion, based on the rates of the two opposite flows, may take place only in the wide section **38** or may begin in the narrow quartz tube **32**. The former option occurs when the diffusion of the  $H_2$  into the tube **32** is hindered by the fast nitrogen stream (100cm/min) mixed with the  $WO_3$  powder, and leads to the formation of  $IF-WS_2$  nanoparticles. The latter case occurs when there occurs some diffusion of  $H_2$  into the tube **32** and leads to the formation of a mixture containing also nanotubes. The resulting powder falls slowly down through the wide part **38** of the tube against the upwardly directed gas stream, until the falling powder is lowered to the narrow part of the tube **36**. High linear rate of the  $N_2/H_2/H_2S$  stream in the narrow part of the tube **36** offers fluidized bed conditions for the falling powder. The filter **40** placed at the end of the constant temperature region at  $830^\circ C$  collects a small part of the powder that dropped on the filter. A very small part of the powder may be swept by the main stream and may be

collected on the additional filter **42** placed outside the tube (i.e., outside the reaction zone) at the end of the constant temperature region (which under the present conditions is at  $830^\circ C$ ) on the way of the gases exit. However, most of the powder is fluidized in the space above the filter **40** in the narrow part of the tube **36**, and stay in the fluidized bed region. It should be understood that, once the gas flow from the bottom of the reactor is terminated, the powder fluidized above the filter also falls onto the filter to be thereby collected.

**[0037]** Fig. 3B illustrates the temperature profile along the reaction path (z-axis) inside the reactor **30**, showing that the temperature during the various process stages is maintained constant.

**[0038]** In this way, more than about 100 grams of closed and hollow  $IF-WS_2$  nanostructures could be synthesized in one batch (ca. 20hr). Thus, the fluidized bed reactor **30** may serve as a prototype for the industrial production of  $IF-WS_2$ .

**[0039]** Fig. 4A show TEM images of  $IF-WS_2$  nanoparticles obtained by the prior art techniques (Fig. 4A(a)) and those obtained by the reactor **20** of the present invention (Fig. 4A(b-f)), wherein Fig. 4A(c,d,e) show the different projections of the same particle by tilting the sample (gride) at an angle of  $(0^\circ \pm 40^\circ)$ , and Fig. 4A(f) shows a rather large  $IF$  (about 400nm). Fig. 4B shows the particles of the prior art technique (Fig. 4B(a)) and of the invented techniques (Fig. 4B(b)) of Fig. 4A in an enlarged scale.

**[0040]** As evident, the shape of the  $IF$  particle of the present invention is very close to a sphere. The electron diffraction pattern of the 400nm- $IF$  particle of Fig. 4A(f), which is shown in the center of the picture, has also proved to be of a fullerene-like structure of the nanoparticle. The particles obtained by the present invention have more perfect spherical shape, than those obtained by the conventional synthetic tools. This stems from the fact that, according to the invention, the reaction takes place in the gas phase, where an isotropic environment for the reaction prevails. Consequently, much larger oxide nanoparticles could be converted into  $IF$  when they flow in the gas stream.

**[0041]** Fig. 5 illustrates the SEM image of the experimental results obtained with the falling bed reactor **20**, showing spherical relatively large  $IF-WS_2$  nanoparticles. The technique of the present invention enables to obtain  $IF$  particles with the size up to  $0.5\mu m$ , compared to  $0.2\mu m$ -particles obtained with the prior art reactor. Additionally, according to the invention, pure  $IF$  phase (with no extra contaminants) is obtained in these processes. Therefore, expensive and time-consuming filtration or purification processes are avoided.

**[0042]** It should be understood that the above-described process of whiskers based nanotubes formation is relevant with respect to the reactor **30** as well. The resulting  $WS_2$  nanotubes obtained by the two reactors **20** and **30** of the present invention have a maximum length of several hundreds  $\mu m$  and a cross-section of

200 nanometer. Fig. 6 illustrates the resulting nanotubes, showing a high resolution TEM image of the WS<sub>2</sub> nanotubes which are multilayer structure, the distance between each two adjacent layers being ca. 6.2 Å.

**[0043]** It should be understood that the use of the vertical posture of the furnace allows addition of the oxide powder into the reactor during the reaction, continuously. The maximum production yield per batch is about 20 grams of a pure *IF*-WS<sub>2</sub> phase by the falling bed reactor and is more than about 100 grams of a pure *IF*-WS<sub>2</sub> phase by the fluidized bed reactor. It should also be understood that the fluidized bed concept lends itself for scale-up and to production of appreciably larger amounts of a pure *IF*-WS<sub>2</sub> phase.

**[0044]** An analysis of the stability of the nanoparticles and nanotubes formed by the present invention reveals that spontaneous transformation of *IF*-WS<sub>2</sub> to 2H has not been observed in any case. One would assume that as the size of the *IF* particles increases, the particles become closer (i.e., less bound) to the 2H particles in their structure. For example, the spacing between the two WS<sub>2</sub> layers (c/2) in the *IF* structures is larger than for the 2H particles. This lattice expansion diminishes with increasing size of the *IF* nanoparticles. It would thus be expected that the larger are the *IF* nanoparticles, the less is the stability of the *IF* to the 2H transformation. Another possible mechanism for the destruction of the formed *IF* nanoparticles under harsh environmental conditions is possible oxidation, which seems more likely than a transformation into 2H platelets. On the other hand, the increase of the *IF*-WS<sub>2</sub> nanoparticle size results in the increase of their oxidation temperature due to the reduced strain in the lattice. Therefore, practically the stability of the larger *IF* to oxidation must be higher.

**[0045]** As mentioned above, the tribological properties of the *IF* nanoparticles are closely related to their structure. In general, spherical *IF* nanoparticles exhibited tribological properties superior to their 2H counterpart, while their performance rapidly deteriorated upon loss of spherical shape. Therefore, the relatively spherical *IF*-WS<sub>2</sub> nanoparticles, which are produced by the falling (fluidized) bed reactor, perform very well as solid lubricants. In addition, the larger *IF* must be a better lubricant in the case of the friction of two matting metal surfaces having higher surface roughness. These hypotheses are substantiated by our recent tribological measurements. For example, it was found that addition of small amount of *IF*-WS<sub>2</sub> to a lubrication fluid leads to a significant improvement of the tribological properties as compared to regular WS<sub>2</sub> (MoS<sub>2</sub>) powders in the same concentration and to a pure fluid lubricant.

**[0046]** Although the invention has been described in conjunction with specific embodiments, it is evident that many alternatives and variations will be apparent to those skilled in the art in light of the foregoing description. Accordingly, the invention is intended to embrace all of the alternatives and variations that fall within the scope of the appended claims.

## Claims

1. An apparatus for the production of inorganic fullerene-like (*IF*) nanoparticles and nanotubes, from metal oxide and reacting gases, the apparatus comprising a chemical reactor, and is associated with a feeding set up and with a temperature control means for controlling the temperature along the reaction path inside the reactor so as to maintain the temperature to be substantially constant, the chemical reactor comprising:
  - (i) a flow line for providing a continuous flow of a mixture of a reacting metal oxide powder and an inert flow assisting gas from the feeding setup into the reaction path, which extends along the reaction path;
  - (ii) at least one inlet port for supplying the reacting gases into the reaction path to flow through said inlet port and along the reaction path, an interaction between the reacting gases and said mixture resulting in the formation of the *IF* nanostructures and nanotubes during the flow of the interacting materials along the reaction path;
  - (iii) at least one outlet port for trapping reacting gases; and
  - (iv) at least one collector located at the opposite end portion of the reactor for collecting the formed *IF*-nanostructures or nanotubes.
2. The apparatus according to claim 1, wherein the feeding set-up comprises a flexible portion thereof, by which it is to be connected to the flow line of the reactor.
3. The apparatus according to claim 1, wherein the feeding set-up comprises two flexibly inter-connected containers aligned along an axis of feeding the mixture, each container having a mesh at its bottom and being mounted on a vibration assembly, the lower one of the containers being connected to the flow line of the reactor.
4. The apparatus according to claim 1, wherein the reaction path extends along a vertical axis of the reactor and the at least one collector is located at the bottom of the reaction path.
5. The apparatus according to claim 1, wherein the at least one inlet port for supplying the reacting gases is located upstream of the reaction path, said flow of the reacting gases towards and through the reaction path being assisted by an inert gas.
6. The apparatus according to claim 1, wherein the at least one inlet port for supplying the reacting gases is located downstream of the reaction path, said flow of the reacting gases towards and through the reac-



tion path being assisted by an inert gas.

7. The apparatus according to claim 1, wherein said metal oxide powder is  $\text{WO}_3$ .
8. The apparatus according to claim 7, wherein said *IF* nanostructures contain hollow cage  $\text{WS}_2$  nanoparticles with fullerene-like structure.
9. The apparatus according to claim 7, wherein said *IF* nanostructures contain a combination of  $\text{WS}_2$  nanoparticles and nanotubes.
10. The apparatus according to claim 7, wherein the temperature within the reaction path is from about  $750^\circ\text{C}$  to about  $850^\circ\text{C}$ .
11. The apparatus according to claim 8, wherein said nanoparticles have a size of up to  $0.5\mu\text{m}$ .
12. The apparatus according to claim 9, wherein the  $\text{WS}_2$  nanotubes have a length of up to a few hundreds  $\mu\text{m}$  and a cross-section of up to 200 nanometer.
13. The apparatus according to claim 1, wherein said chemical reactor is a quartz tube.
14. The apparatus according to claim 1, wherein said metal oxide powder is  $\text{WO}_3$ , said chemical reactor is a quartz tube, and further comprises a lower substantially conical-shape tube extending along a vertical axis between the bottom portion of the reactor and said flow line, such that a distal end of the flow line is located inside an upper wide portion of the lower tube.
15. A synthetic method of production of  $\text{WS}_2$  fullerene-like nanoparticles and nanotubes from tungsten oxide and reacting gases, said method comprising the steps of:
  - (i) providing a chemical reactor defining a reaction path with predetermined temperature conditions along the reaction path;
  - (ii) feeding into the chemical reactor a mixture of a reacting tungsten oxide powder and an inert gas with a predetermined feeding rate, and feeding the reacting gases so as to provide a continuous flow of the mixture and the gases through the reaction path;
  - (iii) maintaining the predetermined temperature to be substantially constant, thereby providing a solid-gas reaction between the reacting gases and the tungsten oxide powder during the flow through and along the reaction path, resulting in the formation of  $\text{WO}_{3-x}$  clusters coated by at least one layer of  $\text{WS}_2$ , and further exchange of

oxides by sulfides in inner parts of the clusters, thereby causing formation of  $\text{WS}_2$  nanostructures; and  
(iv) collecting the formed  $\text{WS}_2$  nanostructures.

16. The method according to claim 15, wherein the feeding rate of the reacting tungsten oxide powder is selected in accordance with the dimensions of the chemical reactor, thereby obtaining the  $\text{WS}_2$  nanostructures in the form of *IF*-nanoparticles, or a combination of *IF*-nanoparticles and nanotubes.
17. The method according to claim 16, wherein the feeding rate ranges between about 3mg/min to about 120mg/min, the reacting gases being  $\text{H}_2$  and  $\text{H}_2\text{S}$ , the concentration of the  $\text{H}_2$  gas being from about 0.2 % (w/v) to about 7 % (w/v) and the concentration of the  $\text{H}_2\text{S}$  gas being from about 0.1 % (w/v) to about 6 % (w/v).
18. The method according to claim 17, wherein the formed nanostructures contain the combination of nanoparticles and nanotubes.
19. The method according to claim 16, wherein the feeding rate ranges between about 3mg/min to about 40mg/min, the reacting gases being  $\text{H}_2$  and  $\text{H}_2\text{S}$  with the concentration of the  $\text{H}_2$  and  $\text{H}_2\text{S}$  gases ranging from about 4 % to about 6 % (w/v).
20. The method according to claim 19, wherein the formed *IF-WS<sub>2</sub>* nanostructures contain the nanoparticles, each having a substantially perfect spherical shape and having a size up to  $0.5\mu\text{m}$ .
21. The method according to claim 16, wherein the feeding rate ranges between about 40mg/min and about 120mg/min, the reacting gases being  $\text{H}_2$  and  $\text{H}_2\text{S}$  with the concentration of the  $\text{H}_2$  and  $\text{H}_2\text{S}$  gases from about 4 % to about 6 % (w/v).
22. The method according to claim 21, wherein the formed  $\text{WS}_2$  nanostructures contain the combination of *IF*-nanoparticles and nanotubes, the nanotubes each having a length of up to a few hundreds  $\mu\text{m}$  and a cross-sectional dimension of up to 200 nanometer.
23. The method according to claim 16, wherein the feeding rate ranges between about 3mg/min to about 40mg/min, the reacting gases being  $\text{H}_2$  and  $\text{H}_2\text{S}$  with the concentration of the  $\text{H}_2$  gas from about 0.2 % to about 4 % and the concentration of the  $\text{H}_2\text{S}$  gas from about 0.2 % to about 4 % (w/v).
24. The method according to claim 23, wherein the formed  $\text{WS}_2$  nanostructures contain the combination of nanoparticles and nanotubes, the nanotubes hav-

ing a length of up to several hundred  $\mu\text{m}$  and a cross-sectional dimension of up to 200 nanometer.

## Patentansprüche

1. Vorrichtung zur Herstellung von anorganischen fullerenartigen (IF-) Nanopartikeln und Nanoröhren aus Metalloxid und Reaktionsgasen, wobei die Vorrichtung einen Chemiereaktor aufweist und einer Einbringeinrichtung und einer Temperaturreguleinrichtung zugeordnet ist, zum Steuern einer Temperatur entlang des Reaktionsweges im Innern des Reaktors, um die Temperatur im wesentlichen konstant zu halten, wobei der Chemiereaktor aufweist:

(i) eine Strömungsleitung zur Bereitstellung einer kontinuierlichen Strömung eines Gemischs aus einem Metalloxid-Reaktionspulver und einem strömungsunterstützenden Inertgas aus der Einbringeinrichtung in den Reaktionsweg, die sich entlang des Reaktionsweges erstreckt;

(ii) mindestens eine Einlaßöffnung zum Einleiten der Reaktionsgase in den Reaktionsweg, so daß diese durch die Einlaßöffnung und entlang des Reaktionsweges strömen, wobei ein Zusammenwirken zwischen den Reaktionsgasen und dem Gemisch zur Bildung der IF-Nanostrukturen und -Nanoröhren während des Strömens der zusammenwirkenden Materialien entlang des Reaktionsweges führt;

(iii) mindestens eine Auslaßöffnung zum Auffangen von Reaktionsgasen; und

(iv) mindestens einen Kollektor, der sich am entgegengesetzten Endabschnitt des Reaktors befindet, zum Sammeln der gebildeten IF-Nanostrukturen oder -Nanoröhren.

2. Vorrichtung nach Anspruch 1, wobei die Einbringeinrichtung einen flexiblen Abschnitt aufweist, durch den sie mit der Strömungsleitung des Reaktors zu verbinden ist.
3. Vorrichtung nach Anspruch 1, wobei die Einbringeinrichtung zwei flexibel miteinander verbundene Behälter aufweist, die entlang einer Gemischeinbringachse ausgerichtet sind, jeder Behälter ein Sieb an seinem Boden hat und auf einer Vibrationsanordnung angeordnet ist und der untere der Behälter mit der Strömungsleitung des Reaktors verbunden ist.
4. Vorrichtung nach Anspruch 1, wobei sich der Reaktionsweg entlang einer vertikalen Achse des Reaktors erstreckt und sich der mindestens eine Kollektor am Boden des Reaktorwegs befindet.
5. Vorrichtung nach Anspruch 1, wobei sich die mindestens eine Einlaßöffnung zum Einleiten des Reaktionsgases in Strömungsrichtung vor dem Reaktionsweg befindet und die Strömung der Reaktionsgase zum und durch den Reaktionsweg durch ein Inertgas unterstützt wird.

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6. Vorrichtung nach Anspruch 1, wobei sich mindestens eine Einlaßöffnung zum Einleiten der Reaktionsgase in Strömungsrichtung hinter dem Reaktionsweg befindet und die Strömung der Reaktionsgase zum und durch den Reaktionsweg durch ein Inertgas unterstützt wird.

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7. Vorrichtung nach Anspruch 1, wobei das Metalloxidpulver  $\text{WO}_3$  ist.

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8. Vorrichtung nach Anspruch 7, wobei die IF-Nanostrukturen Hohlmantel- $\text{WS}_2$ -Nanopartikel mit fullerenartiger Struktur enthalten.

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9. Vorrichtung nach Anspruch 7, wobei die IF-Nanostrukturen eine Kombination aus  $\text{WS}_2$ -Nanopartikeln und -Nanoröhren enthalten.

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10. Vorrichtung nach Anspruch 7, wobei die Temperatur innerhalb des Reaktionsweges von etwa  $750^\circ\text{C}$  bis etwa  $850^\circ\text{C}$  beträgt.

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11. Vorrichtung nach Anspruch 8, wobei die Nanopartikel eine Größe von bis zu  $0,5 \mu\text{m}$  haben.

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12. Vorrichtung nach Anspruch 9, wobei die  $\text{WS}_2$ -Nanoröhren eine Länge von bis zu wenigen hundert  $\mu\text{m}$  und einen Querschnitt von bis zu 200 nm haben.

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13. Vorrichtung nach Anspruch 1, wobei der Chemiereaktor eine Quarzröhre ist.

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14. Vorrichtung nach Anspruch 1, wobei das Metalloxidpulver  $\text{WO}_3$  ist, der Chemiereaktor eine Quarzröhre ist und ferner eine untere im wesentlichen konische Röhre aufweist, die sich entlang einer vertikalen Achse zwischen dem Bodenabschnitt des Reaktors und der Strömungsleitung erstreckt, so daß sich ein distales Ende der Strömungsleitung im Innern eines oberen breiten Abschnitts der unteren Röhre befindet.

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15. Syntheseverfahren zur Herstellung von fullerenartigen  $\text{WS}_2$ -Nanopartikeln und -Nanoröhren aus Wolframoxid und Reaktionsgasen, wobei das Verfahren die Schritte aufweist:

(i) Bereitstellen eines Chemiereaktors, der einen Reaktionsweg mit vorbestimmten Temperaturbedingungen entlang des Reaktionsweges definiert;

(ii) Einbringen eines Gemischs aus einem Wolframoxid-Reaktionspulver und einem Inertgas

- mit einer vorbestimmten Einbringrate in den Chemiereaktor und Einbringen der Reaktionsgase, um eine kontinuierliche Strömung des Gemischs und der Gase durch den Reaktorweg zu ermöglichen;
- (iii) Halten der vorbestimmten Temperatur, so daß diese im wesentlichen konstant ist, wodurch eine Feststoff-Gas-Reaktion zwischen den Reaktionsgasen und dem Wolframoxidpulver während des Strömens durch und entlang des Reaktionsweges erfolgt, was zur Bildung von mit mindestens einer  $WS_2$ -Schicht überzogenen  $WO_{3-x}$ -Clustern und zu weiterem Austausch von Oxiden durch Sulfide in inneren Teilen der Cluster führt, wodurch die Bildung von  $WS_2$ -Nanostrukturen bewirkt wird; und
- (iv) Sammeln der gebildeten  $WS_2$ -Nanostrukturen.
16. Verfahren nach Anspruch 15, wobei die Einbringrate des Wolframoxid-Reaktionspulvers entsprechend den Abmessungen des Chemiereaktors gewählt wird, wodurch die  $WS_2$ -Nanostrukturen in Form von IF-Nanopartikeln oder einer Kombination aus IF-Nanopartikeln und -Nanoröhren gewonnen werden.
17. Verfahren nach Anspruch 16, wobei die Einbringrate von etwa 3 mg/min bis etwa 120 mg/min reicht, die Reaktionsgase  $H_2$  und  $H_2S$  sind, wobei die Konzentration des  $H_2$ -Gases von etwa 0,2 Gew./Vol.-% bis etwa 7 Gew./Vol.-% beträgt und die Konzentration des  $H_2S$ -Gases etwa 0,1 Gew./Vol.-% bis etwa 6 Gew./Vol.-% beträgt.
18. Verfahren nach Anspruch 17, wobei die gebildeten Nanostrukturen die Kombination aus Nanopartikeln und Nanoröhren enthalten.
19. Verfahren nach Anspruch 16, wobei die Einbringrate von etwa 3 mg/min bis etwa 40 mg/min reicht, die Reaktionsgase  $H_2$  und  $H_2S$  sind, wobei die Konzentration der  $H_2$ - und  $H_2S$ -Gase von etwa 4 Gew./Vol.-% bis etwa 6 Gew./Vol.-% reicht.
20. Verfahren nach Anspruch 19, wobei die gebildeten IF- $WS_2$ -Nanostrukturen die Nanopartikel enthalten, die jeweils eine im wesentlichen ideale Kugelform und eine Größe bis zu 0,5  $\mu m$  haben.
21. Verfahren nach Anspruch 16, wobei die Einbringrate zwischen etwa 40 mg/min und etwa 120 mg/min liegt, die Reaktionsgase  $H_2$  und  $H_2S$  sind und die Konzentration der  $H_2$ - und  $H_2S$ -Gase von etwa 4 Gew./Vol.-% bis etwa 6 Gew./Vol.-% beträgt.
22. Verfahren nach Anspruch 21, wobei die gebildeten  $WS_2$ -Nanostrukturen die Kombination aus IF-Nanopartikeln und -Nanoröhren enthalten und die Nano-

röhren jeweils eine Länge von bis zu wenigen hundert  $\mu m$  und eine Querschnittsabmessung von bis zu 200 nm haben.

23. Verfahren nach Anspruch 16, wobei die Einbringrate von etwa 3 mg/min bis etwa 40 mg/min reicht, die Reaktionsgase  $H_2$  und  $H_2S$  sind, wobei die Konzentration des  $H_2$ -Gases von etwa 0,2 % bis etwa 4 % und die Konzentration des  $H_2S$ -Gases von etwa 0,2 Gew./Vol.-% bis etwa 4 Gew./Vol.-% reicht.
24. Verfahren nach Anspruch 23, wobei die gebildeten  $WS_2$ -Nanostrukturen die Kombination aus Nanopartikeln und Nanoröhren enthalten und die Nanoröhren eine Länge von bis zu mehreren hundert  $\mu m$  und eine Querschnittsabmessung von bis zu 200 nm haben.

## 20 Revendications

- Appareil destiné à la production de nanoparticules et nanotubes inorganiques de type fullerène (IF), à partir d'oxyde de métal et de gaz de réaction, l'appareil comprenant un réacteur chimique, et est associé à une installation d'alimentation et un moyen de régulation de température pour réguler la température le long du chemin de réaction à l'intérieur du réacteur de sorte à maintenir la température sensiblement constante, le réacteur chimique comprenant :
  - une conduite d'écoulement pour fournir un écoulement continu d'un mélange d'une poudre d'oxyde de métal de réaction et d'un gaz inerte facilitant l'écoulement à partir de l'installation d'alimentation dans le chemin de réaction, qui s'étend le long du chemin de réaction ;
  - au moins un orifice d'entrée pour fournir les gaz de réaction dans le chemin de réaction pour s'écouler à travers ledit orifice d'entrée et le long du chemin de réaction, une interaction entre les gaz de réaction et ledit mélange entraînant la formation des nanostructures et nanotubes IF pendant l'écoulement des matériaux d'interaction le long du chemin de réaction ;
  - au moins un orifice de sortie pour piéger les gaz de réaction ; et
  - au moins un collecteur situé au niveau de la portion d'extrémité opposée du réacteur pour collecter les nanostructures ou nanotubes IF formés.
- Appareil selon la revendication 1, dans lequel l'installation d'alimentation comprend une portion flexible de celle-ci, par laquelle elle doit être reliée à la conduite d'écoulement du réacteur.

3. Appareil selon la revendication 1, dans lequel l'installation d'alimentation comprend deux conteneurs interconnectés de manière flexible alignés le long d'un axe d'alimentation du mélange, chaque conteneur ayant un tamis au niveau de son fond et étant monté sur un ensemble de vibration, le plus bas des conteneurs étant relié à la conduite d'écoulement du réacteur. 5
4. Appareil selon la revendication 1, dans lequel le chemin de réaction s'étend le long d'un axe vertical du réacteur et le au moins un collecteur est situé au fond du chemin de réaction. 10
5. Appareil selon la revendication 1, dans lequel le au moins un orifice d'entrée pour fournir les gaz de réaction est situé en amont du chemin de réaction, ledit écoulement des gaz de réaction vers et à travers le chemin de réaction étant facilité par un gaz inerte. 15
6. Appareil selon la revendication 1, dans lequel le au moins un orifice d'entrée pour fournir les gaz de réaction est situé en aval du chemin de réaction, ledit écoulement des gaz de réaction vers et à travers le chemin de réaction étant facilité par un gaz inerte. 20
7. Appareil selon la revendication 1, dans lequel ladite poudre d'oxyde de métal est le  $WO_3$ . 25
8. Appareil selon la revendication 7, dans lequel lesdites nanostructures IF contiennent des nanoparticules de  $WS_2$  à cage creuse avec une structure de type fullerène. 30
9. Appareil selon la revendication 7, dans lequel lesdites nanostructures IF contiennent une combinaison de nanoparticules et de nanotubes de  $WS_2$ . 35
10. Appareil selon la revendication 7, dans lequel la température au sein du chemin de réaction est d'environ  $750\text{ }^{\circ}\text{C}$  à environ  $850\text{ }^{\circ}\text{C}$ . 40
11. Appareil selon la revendication 8, dans lequel lesdites nanoparticules ont une taille allant jusqu'à  $0,5\text{ }\mu\text{m}$ . 45
12. Appareil selon la revendication 9, dans lequel les nanotubes de  $WS_2$  ont une longueur allant jusqu'à quelques centaines de  $\mu\text{m}$  et une section transversale allant jusqu'à 200 nanomètres. 50
13. Appareil selon la revendication 1, dans lequel ledit réacteur chimique est un tube de quartz.
14. Appareil selon la revendication 1, dans lequel ladite poudre d'oxyde de métal est le  $WO_3$ , ledit réacteur chimique est un tube de quartz, et comprend en outre un tube inférieur sensiblement de forme conique s'étendant le long d'un axe vertical entre la portion inférieure du réacteur et ladite conduite d'écoulement, de sorte qu'une extrémité distale de la conduite d'écoulement est située à l'intérieur d'une portion large supérieure du tube inférieur.
15. Procédé synthétique de production de nanoparticules et nanotubes de type fullerène de  $WS_2$  à partir d'oxyde de tungstène et de gaz de réaction, ledit procédé comprenant les étapes consistant à :
  - (i) fournir un réacteur chimique définissant un chemin de réaction avec des conditions de température prédéterminées le long du chemin de réaction ;
  - (ii) alimenter le réacteur chimique en un mélange d'une poudre d'oxyde de tungstène de réaction et d'un gaz inerte avec une vitesse d'alimentation prédéterminée, et alimenter en gaz de réaction de sorte à fournir un écoulement continu du mélange et des gaz à travers le chemin de réaction ;
  - (iii) maintenir la température prédéterminée sensiblement constante, fournissant ainsi une réaction solide-gaz entre les gaz de réaction et la poudre d'oxyde de tungstène pendant l'écoulement à travers et le long du chemin de réaction, entraînant la formation d'agrégats de  $WO_{3-x}$  revêtus d'au moins une couche de  $WS_2$ , et l'échange ultérieur d'oxydes par sulfures dans les parties internes des agrégats, provoquant ainsi la formation de nanostructures de  $WS_2$  ; et
  - (iv) collecter les nanostructures de  $WS_2$  formées.
16. Procédé selon la revendication 15, dans lequel la vitesse d'alimentation de la poudre d'oxyde de tungstène de réaction est sélectionnée selon les dimensions du réacteur chimique, obtenant ainsi les nanostructures de  $WS_2$  sous la forme de nanoparticules IF, ou une combinaison de nanoparticules et de nanotubes IF.
17. Procédé selon la revendication 16, dans lequel la vitesse d'alimentation est comprise entre environ 3 mg/min et environ 120 mg/min, les gaz de réaction étant  $H_2$  et  $H_2S$ , la concentration du gaz  $H_2$  étant d'environ 0,2 % (p/v) à environ 7 % (p/v) et la concentration du gaz  $H_2S$  étant d'environ 0,1 % (p/v) à environ 6 % (p/v) .
18. Procédé selon la revendication 17, dans lequel les nanostructures formées contiennent la combinaison de nanoparticules et de nanotubes.
19. Procédé selon la revendication 16, dans lequel la vitesse d'alimentation est comprise entre environ 3 mg/min et environ 40 mg/min, les gaz de réaction

étant  $H_2$  et  $H_2S$  avec la concentration des gaz  $H_2$  et  $H_2S$  comprise entre environ 4 % et environ 6 % (p/v).

20. Procédé selon la revendication 19, dans lequel les nanostructures de  $WS_2$ -IF formées contiennent les nanoparticules, chacune ayant une forme sphérique sensiblement parfaite et ayant une taille allant jusqu'à  $0,5 \mu m$ , 5
21. Procédé selon la revendication 16, dans lequel la vitesse d'alimentation est comprise entre environ 40 mg/min et environ 120 mg/min, les gaz de réaction étant  $H_2$  et  $H_2S$  avec la concentration des gaz  $H_2$  et  $H_2S$  d'environ 4 % à environ 6 % (p/v). 10 15
22. Procédé selon la revendication 21, dans lequel les nanostructures de  $WS_2$  formées contiennent la combinaison de nanoparticules et de nanotubes IF, les nanotubes ayant chacun une longueur allant jusqu'à quelques centaines de  $\mu m$  et une dimension en coupe transversale allant jusqu'à 200 nanomètres. 20
23. Procédé selon la revendication 16, dans lequel la vitesse d'alimentation est comprise entre environ 3 mg/min et environ 40 mg/min, les gaz de réaction étant  $H_2$  et  $H_2S$  avec la concentration du gaz  $H_2$  d'environ 0,2 % à environ 4 % et la concentration du gaz  $H_2S$  d'environ 0,2 % à environ 4 % (p/v). 25
24. Procédé selon la revendication 23, dans lequel les nanostructures de  $WS_2$  formées contiennent la combinaison de nanoparticules et de nanotubes, les nanotubes ayant une longueur allant jusqu'à plusieurs centaines de  $\mu m$  et une dimension en coupe transversale allant jusqu'à 200 nanomètres. 30 35

40

45

50

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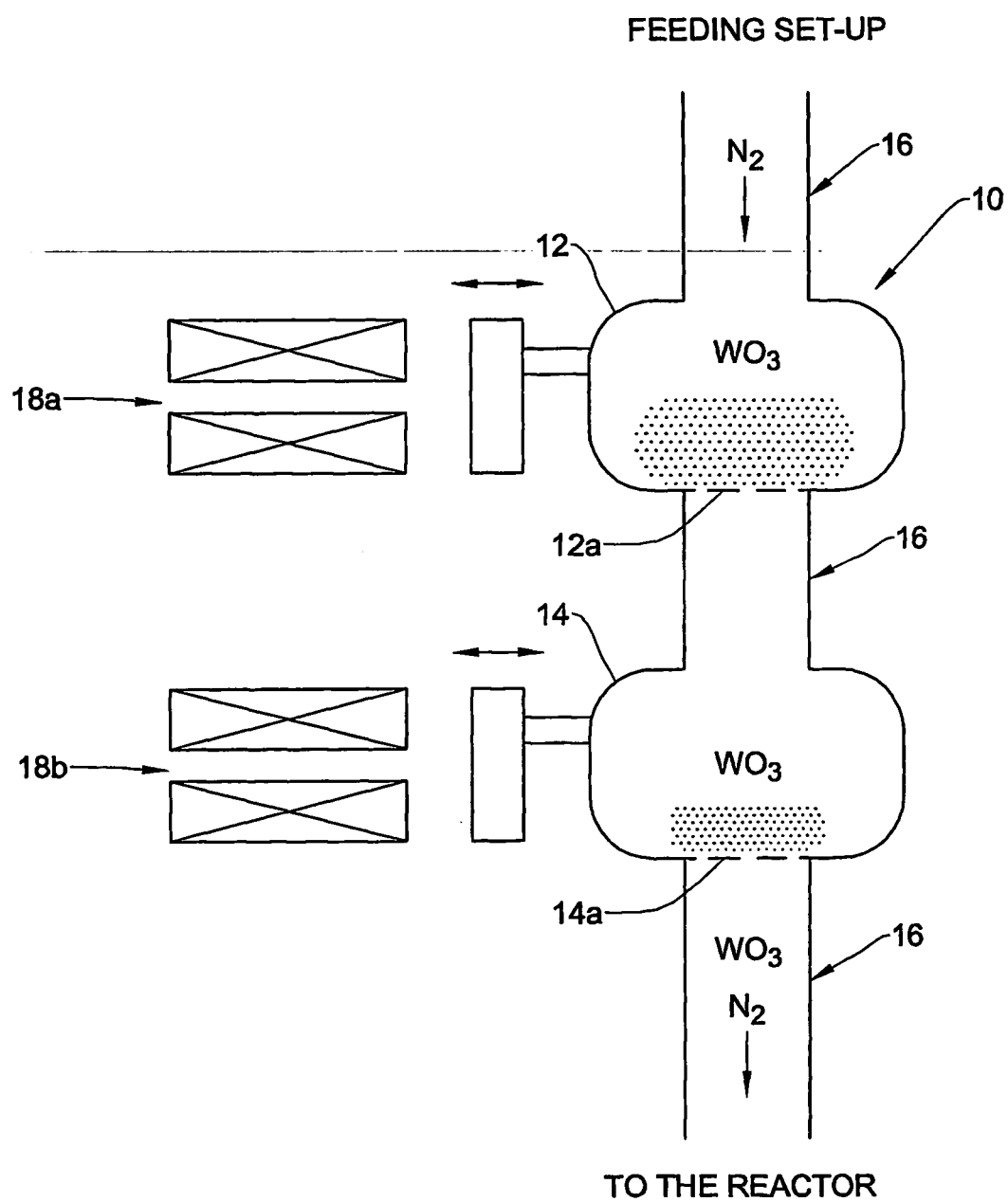


FIG. 1

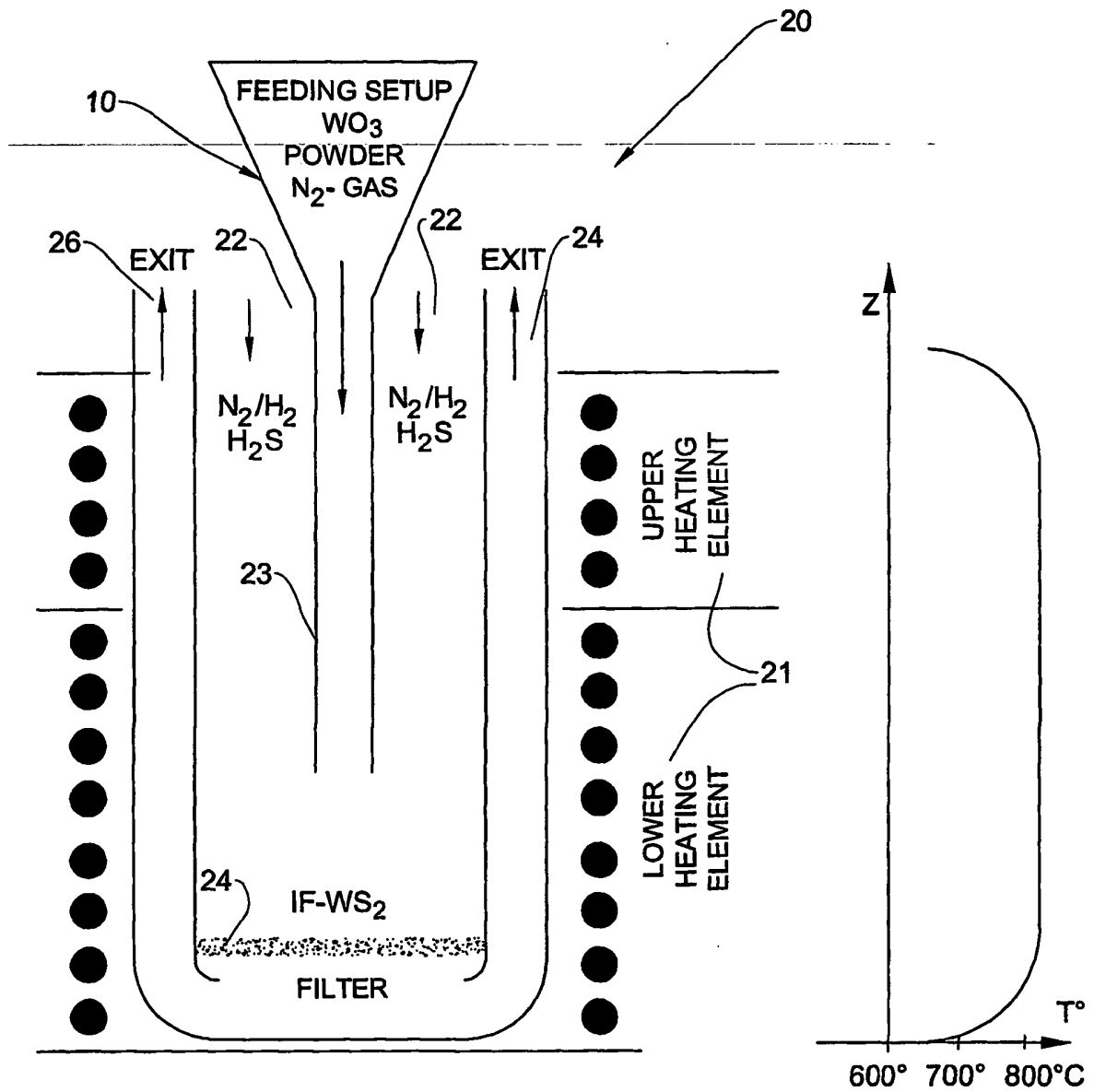


FIG. 2A

FIG. 2B

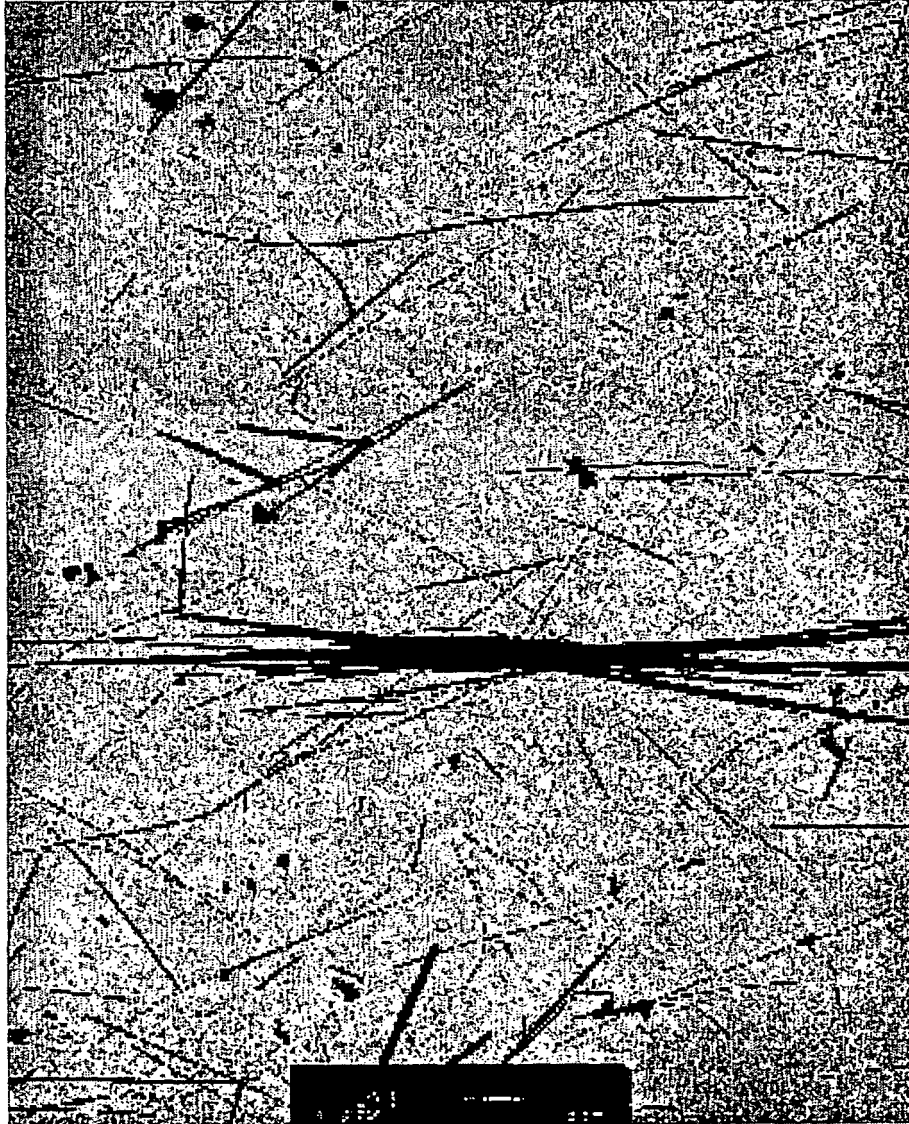


FIG. 2C



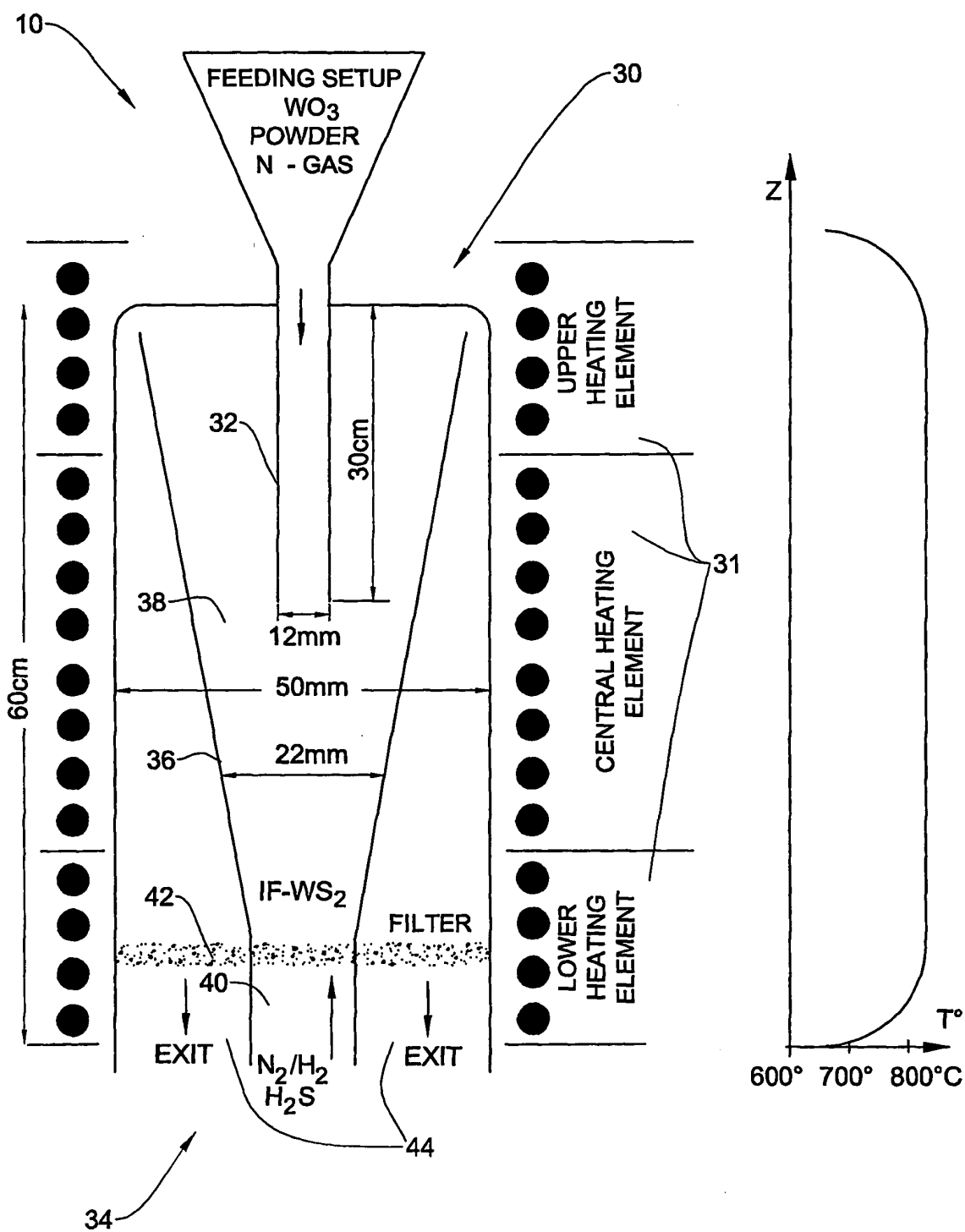


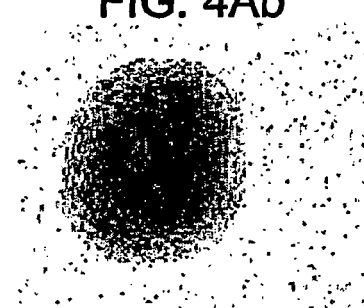
FIG. 3A

FIG. 3B

FIG. 4Aa

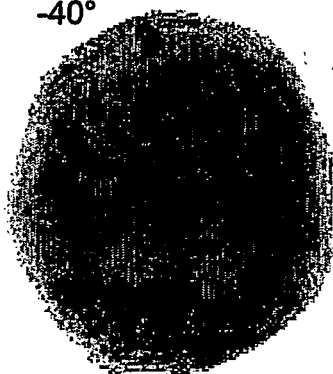


FIG. 4Ab

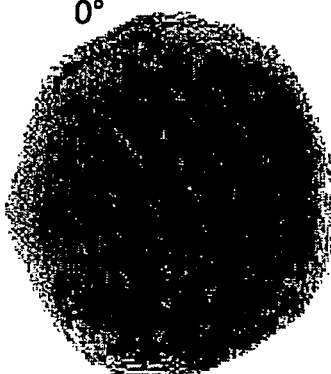


100nm

-40°



0°



+40°

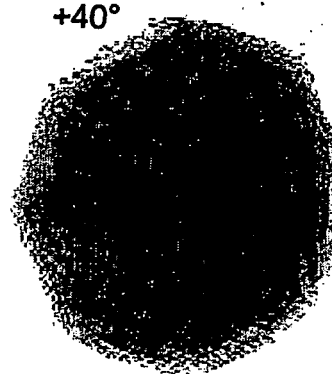


FIG. 4Ac

FIG. 4Ad

FIG. 4Ae

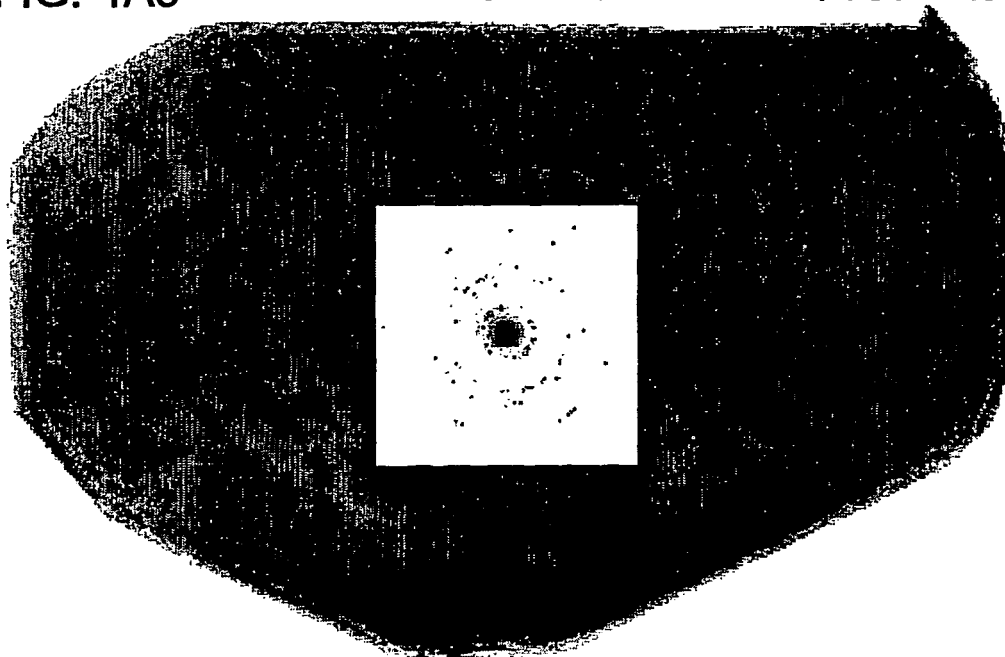


FIG. 4Af



FIG. 4Ba

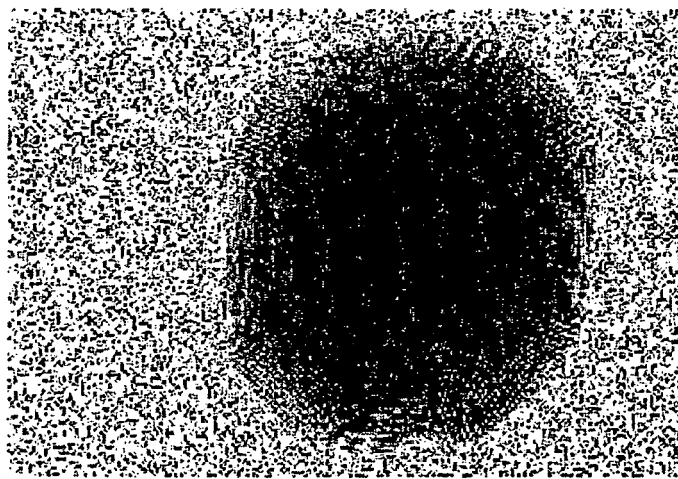


FIG. 4Bb

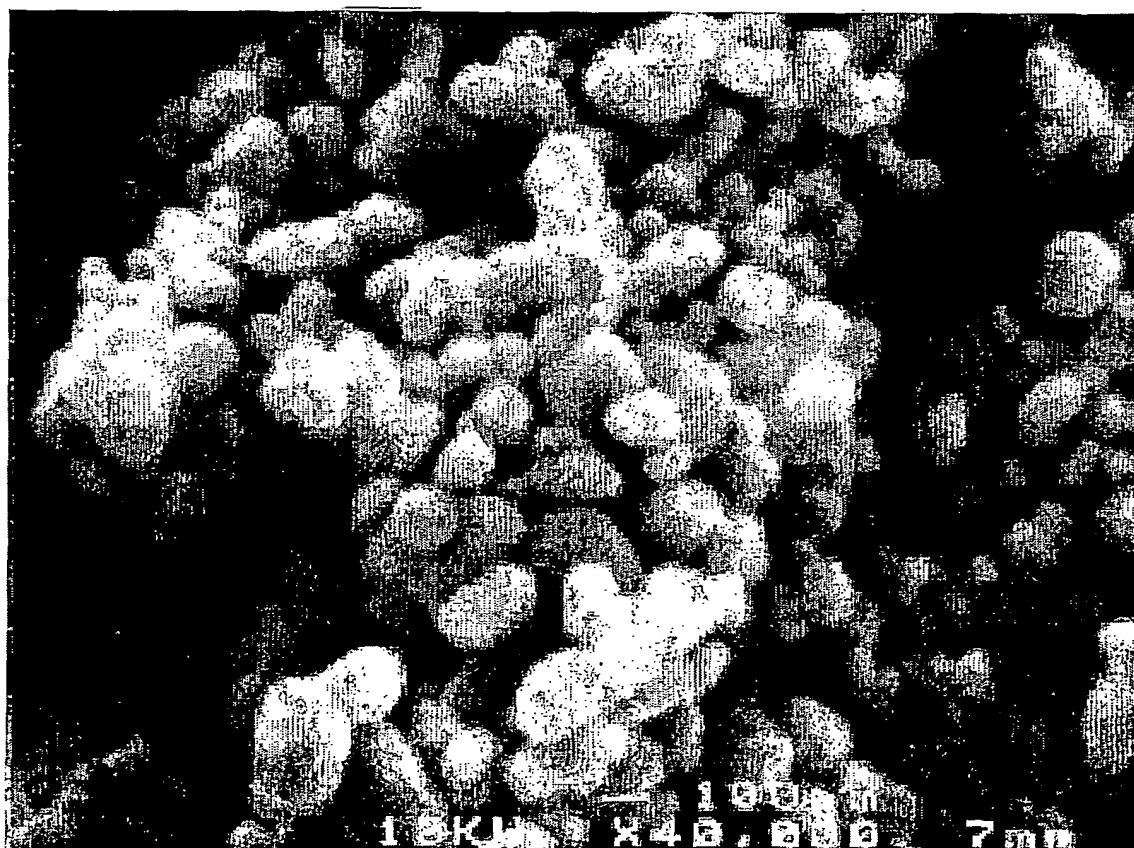


FIG. 5

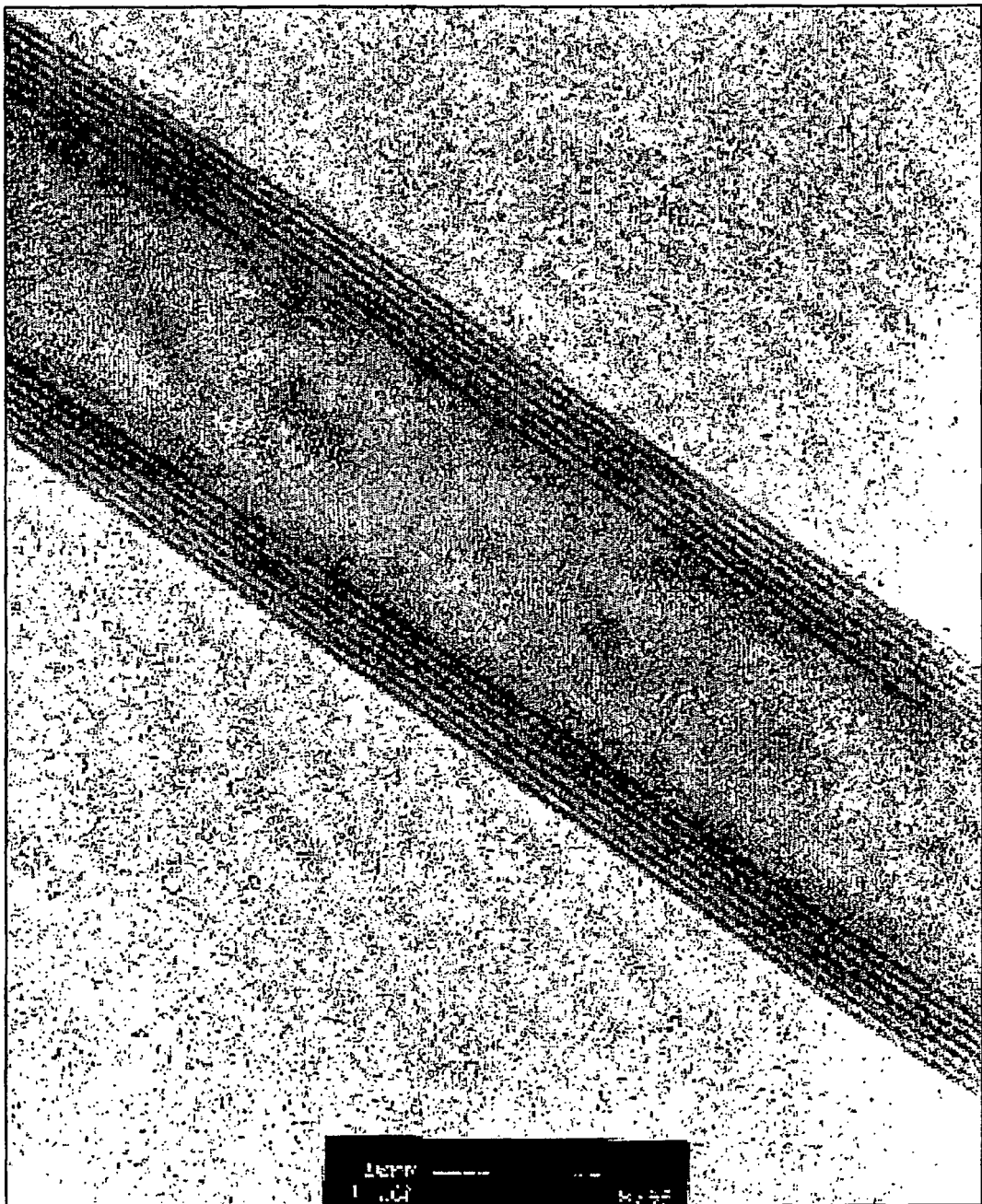


FIG. 6