## Journal of Materials Chemistry

Cite this: J. Mater. Chem., 2012, 22, 21998

www.rsc.org/materials

**PAPER** 

# The GO/rGO-Fe<sub>3</sub>O<sub>4</sub> composites with good water-dispersibility and fast magnetic response for effective immobilization and enrichment of biomolecules<sup>†</sup>

Gong Cheng, abd Yan-Lin Liu, da Zhi-Gang Wang, dd Ji-Lin Zhang, be De-Hui Sun and Jia-Zuan Ni abd Jia-Zuan Ni

Received 8th June 2012, Accepted 24th August 2012

DOI: 10.1039/c2jm33695b

The graphene oxide (GO)–Fe<sub>3</sub>O<sub>4</sub> and the reduced graphene oxide (rGO)–Fe<sub>3</sub>O<sub>4</sub> composites with good water dispersibility, high affinity and rapid magnetic response have been prepared *via* a facile grafting method. They can be used for the effective immobilization of proteins and the enrichment of peptides, respectively. By taking advantage of the high loading capacity and the abundant hydrophilic groups of the GO–Fe<sub>3</sub>O<sub>4</sub> composites, they can be applied to immobilize proteins. The amount of loading of protein (BSA) on GO–Fe<sub>3</sub>O<sub>4</sub> is as high as 294.54 mg g<sup>-1</sup>. The rGO–Fe<sub>3</sub>O<sub>4</sub> composites can be used to enrich the low-concentration peptides conveniently due to their high surface areas, special structures and strong magnetism. Nineteen target peptides with the sequence coverage of 21% can be enriched and detected from the highly diluted digest of BSA (5 fmol  $\mu$ L<sup>-1</sup>). These results reveal that the prepared GO/rGO–Fe<sub>3</sub>O<sub>4</sub> composites have potential application as a carrier for biomolecule immobilization, enrichment, and separation.

#### Introduction

As a new class of two-dimensional carbon nanostructure, graphene has received increasing attention in various fields due to its attractive electronic, catalytic, mechanical, optical, and magnetic properties as well as available hydrophobicity. <sup>1-5</sup> Notably, the large delocalized  $\pi$ -electron system and ultrahigh specific surface area of graphene endow it with a strong affinity and high loading capacity for hydrophobic biomolecules and carbon-based ring structures. <sup>6-9</sup> Graphene oxide (GO), the oxidized derivative of graphene, is highly hydrophilic and water soluble because it possesses various functional groups such as epoxide, carboxyl, and hydroxyl. They can also undergo covalent, electrostatic, or hydrogen bonding for functional modification, which makes them more suitable for bio-application.

Recently, graphene and graphene oxide were developed as fascinating and promising candidate materials in biomedical applications such as drug delivery carriers of high efficiency, purification and detection of biomolecules, immobilization of enzymes, and enrichment of proteins. <sup>10–13</sup> For example, Tang and

co-workers14 demonstrated that graphene can be used as an ultrahigh efficiency preconcentration and detection platform for DNA oligomers for direct mass spectrometry (MS) analysis. Furthermore, the aptamer-conjugated graphene oxide composites were prepared and used as an affinity extraction and detection platform for analytes from complex biological media.<sup>15</sup> Despite these developing achievements, the direct use of graphene or graphene oxide as adsorbents for bio-application has several disadvantages. First, it is difficult to collect the small graphene or graphene oxide sheets from a well-dispersed solution via filtration and centrifugation. Second, the direct separation of graphene or graphene oxide sheets easily results in the irreversible aggregation of them, which may reduce the adsorption capacity and the adsorption efficiency. Therefore, optimizing the surface properties of the graphene or graphene oxide sheets by further functionalization for convenient separation and high adsorption efficiency is required.

As a promising separation tool in bio-application, the Fe<sub>3</sub>O<sub>4</sub> particles-based magnetic isolation has become an effective separation technique. <sup>16–19</sup> Recently, some graphene–Fe<sub>3</sub>O<sub>4</sub> hybrid materials have been explored for energy storage, water treatment, preconcentration of small molecules and drug delivery. <sup>20–23</sup> However, there still exist the following problems which are not in favor of the bio-applications: (1) poor control over the size, size distribution, and location of magnetic Fe<sub>3</sub>O<sub>4</sub> nanoparticles on the graphene oxide or graphene sheets; (2) they may agglomerate and gradually precipitate from their aqueous suspension due to the poor water dispersibility of the hydrophobic graphene. <sup>24,25</sup> Although some polymer-additives were

<sup>&</sup>quot;Key Laboratory of Marine Bioresources and Ecology, College of Life Science, Shenzhen University, Shenzhen 518060, China

bState Key Laboratory of Rare Earth Resource Utilization, Changchun Institute of Applied Chemistry, Chinese Academy of Sciences, Changchun 130022, China. E-mail: zjl@ciac.jl.cn

<sup>&</sup>lt;sup>c</sup>Changchun Institute Technology, Changchun 130012, China

<sup>&</sup>lt;sup>d</sup>Graduate University of the Chinese Academy of Sciences, Beijing 100049, PR China

<sup>†</sup> Electronic supplementary information (ESI) available. See DOI: 10.1039/c2jm33695b

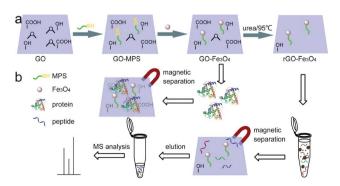
used to produce water-dispersible graphene–Fe<sub>3</sub>O<sub>4</sub> hybrid materials,<sup>26,27</sup> they could be undesirable for practical applications. Additionally, the graphene-based magnetic hybrid materials with good water dispersibility for effective protein immobilization and enrichment of low-concentration peptides as well as easy magnetic separation have not yet been reported.

In this study, the GO/rGO-Fe<sub>3</sub>O<sub>4</sub> composites with good water dispersibility and fast magnetic response have been prepared via a facile grafting method and applied for biomolecule immobilization and enrichment. As shown in Scheme 1a, the GO was first grafted by (3-mercaptopropyl)trimethoxysilane (MPS) with mercapto group, and then the spherical Fe<sub>3</sub>O<sub>4</sub> particles with different sizes were controlled to covalently attach on the GO nanosheets. The obtained product is denoted as GO-Fe<sub>3</sub>O<sub>4</sub>. After the reduction of GO-Fe<sub>3</sub>O<sub>4</sub> in alkaline aqueous solution under mild conditions, the water-dispersible reduced graphene oxide-Fe<sub>3</sub>O<sub>4</sub> composites (denoted as rGO-Fe<sub>3</sub>O<sub>4</sub>) were obtained. The GO/rGO-Fe<sub>3</sub>O<sub>4</sub> composites with good waterdispersity and fast magnetic response combine the merits of the easy separation of Fe<sub>3</sub>O<sub>4</sub> and high affinity of graphene and graphene oxide. These composites have been demonstrated to be a potential platform for protein immobilization and peptide enrichment. The prepared GO-Fe<sub>3</sub>O<sub>4</sub> composites can be used as effective adsorbents for protein immobilization by virtue of their high loading capacity and the abundant hydrophilic groups, while rGO-Fe<sub>3</sub>O<sub>4</sub> composites with high surface areas and hydrophobicity possess great potential for peptide enrichment.

#### **Experimental**

#### Materials

Graphite powder, urea, ammonium bicarbonate (NH<sub>4</sub>HCO<sub>3</sub>), ferric chloride hexahydrate (FeCl<sub>3</sub>·6H<sub>2</sub>O), ethylene glycol (EG), diethylene glycol (DEG), ethylenediamine (EN) and sodium acetate (NaAc) were purchased from China National Pharmaceutical Group Corporation (Shanghai, China). (3-Mercaptopropyl)trimethoxysilane (98%, MPS), polyethylene glycol ( $M_{\rm w}$  1000, PEG), acetonitrile (ACN), 2,5-dihydroxybenzoicacid (2,5-DHB), dithiothreitol (DTT) and bovine serum albumin (98%, BSA) were purchased from Aladdin (Shanghai, China). Trypsin (from bovine pancreas, TPCK treated) was purchased from Sigma-Aldrich (St. Louis, MO, USA). Iodoacetamide



Scheme 1 Schematic illustration of the synthesis strategy of the GO– $Fe_3O_4$  and rGO– $Fe_3O_4$  composites (a), and their application in protein immobilization and peptide enrichment (b).

(IAA) was purchased from Alfa Aesar (USA). K<sub>2</sub>S<sub>2</sub>O<sub>8</sub>, P<sub>2</sub>O<sub>5</sub>, KMnO<sub>4</sub>, H<sub>2</sub>O<sub>2</sub> (30%), H<sub>2</sub>SO<sub>4</sub>, HCl and ethanol were all analytical grade reagents and were obtained from Beijing Chemical Regent Co. Ltd. (Beijing, China).

#### Preparation of GO and GO-MPS

Graphene oxide (GO) was synthesized from natural graphite powder by a modified Hummers method. <sup>28,29</sup> For the preparation of mercapto group-functionalized GO, 15 mg GO was dispersed into 20 mL ethanol *via* ultrasonication for 30 min. Then, the dispersion system was bubbled with nitrogen for 30 min under mechanical agitation. Subsequently, 0.25 mL MPS was slowly added into the dispersion system under nitrogen atmosphere, and then the system was mechanically stirred for 12 h at 70 °C in a water bath. The obtained GO–MPS product was separated by centrifugation, and then washed with ethanol three times to remove the unreacted MPS.

### Preparation of superparamagnetic Fe<sub>3</sub>O<sub>4</sub> particles with different sizes

For the synthesis of 60 nm Fe $_3$ O $_4$  nanoparticles, FeCl $_3$ ·6H $_2$ O (2.50 g) was injected into a mixture containing 8.9 mL EN and 80 mL EG under magnetic stirring, followed by the addition of NaAc (5.00 g). The mixture was transferred to a Teflon-lined stainless-steel autoclave and sealed, to heat at 200 °C for 6 h. The produced magnetite nanoparticles were washed with deionized water, and separated using a magnet, and finally dried in an oven at 80 °C for 12 h. Synthesis of 200 nm Fe $_3$ O $_4$  nanoparticle is similar to the above synthetic process except FeCl $_3$ ·6H $_2$ O (2.70 g), EN (10 mL) and additive PEG (2.00 g) were used. 500 nm Fe $_3$ O $_4$  microspheres were prepared following a literature method reported previously.

#### Preparation of GO-Fe<sub>3</sub>O<sub>4</sub> and rGO-Fe<sub>3</sub>O<sub>4</sub> composites

The GO–Fe<sub>3</sub>O<sub>4</sub> composites were prepared by the following method. 15 mg of the obtained Fe<sub>3</sub>O<sub>4</sub> particles were pretreated with 25 mL 0.1 M hydrochloric acid *via* ultrasonication for 30 min. Then, the particles were separated *via* a magnet and washed with deionized water and ethanol 3 times. The above particles and GO–MPS were re-dispersed into 15 mL ethanol *via* ultrasonication for 30 min. The particle suspension was added dropwise into the GO–MPS solution under agitation. The mixture was further stirred for 24 h at room temperature. Finally, the GO–Fe<sub>3</sub>O<sub>4</sub> particles were collected by a magnet, and then washed with deionized water and ethanol several times.

The rGO–Fe<sub>3</sub>O<sub>4</sub> composites were prepared *via* reducing the GO–Fe<sub>3</sub>O<sub>4</sub> composites in alkaline aqueous solution under mild conditions.<sup>31,32</sup> In brief, the as-prepared GO–Fe<sub>3</sub>O<sub>4</sub> composites were well-dispersed into 30 mL deionized water, and the pH value of the GO–Fe<sub>3</sub>O<sub>4</sub> dispersion solution was adjusted to 8.0 using dilute ammonia solution. Subsequently, 1.0 g urea was added into the solution mixture under agitation, and then the solution mixture was heated to 95 °C. After stirring for 48 h, the rGO–Fe<sub>3</sub>O<sub>4</sub> composites were collected by magnetic separation, and washed with deionized water several times.

#### Tryptic digest of proteins

1 mg protein (BSA) was dissolved in 0.5 mL 50 mM  $NH_4HCO_3$  solution containing urea (4.0 mmol, 8 M). 50  $\mu L$  100 mM DTT solution was added into the protein solution, and then the system was maintained at 60 °C for 1 h. After that, 50  $\mu L$  of 200 mM IAA solution was added into the above solution, followed by shaking for 2.5 h at room temperature. After denaturation by the above step, trypsin was added into the solution with a molar ratio of 50 : 1 (BSA : trypsin), and the solution was incubated at 18 h at 37 °C. Finally, the obtained tryptic digests were diluted to the target concentration.

#### Adsorption of biomolecules by GO-Fe<sub>3</sub>O<sub>4</sub> composites

The GO–Fe<sub>3</sub>O<sub>4</sub> composites (2 mg mL<sup>-1</sup>) were dispersed into a phosphate buffer (PBS) prior to adsorption. 0.5 mL of the GO–Fe<sub>3</sub>O<sub>4</sub> suspended solution (2 mg mL<sup>-1</sup>) was mixed with 0.5 mL of the BSA solution in PBS with designed concentrations, followed by shaking for three hours so as to ensure reaching equilibration. The supernatant was collected after magnetic separation. The adsorption curves were obtained by measuring the solution absorbence at 280 nm before and after the treatment using UV-vis spectroscopy and a calibration curve. The amount of the adsorbed biomolecules was calculated using the following equation:

$$Q_{\rm e} = \frac{(C_0 - C_{\rm e})V}{m} \tag{1}$$

where  $Q_e$  is the concentration of the adsorbed biomolecules (mg g<sup>-1</sup>),  $C_0$  and  $C_e$  are the initial and the equilibrium concentrations of protein, respectively. V is the volume of the solution, and m is the weight of the added GO–Fe<sub>3</sub>O<sub>4</sub> composites.

#### **Enrichment of low-abundance peptides**

 $1000~\mu L$  (5 fmol  $\mu L^{-1}$ ) BSA digest was mixed with  $10~\mu L$  of 35 mg mL<sup>-1</sup> Fe<sub>3</sub>O<sub>4</sub> or rGO–Fe<sub>3</sub>O<sub>4</sub> composites suspended solution and then shaken for 15 min. Subsequently, the composites trapped target peptides were collected by magnetic separation. Then, the enriched peptides were eluted with 5  $\mu L$  of elution solution containing 60% ACN and 1.0% TFA, and the supernatant was collected by magnetic separation. The above products obtained from the elution step were dissolved in 50% (v/v) aqueous acetonitrile solution (2  $\mu L$ ) containing DHB (20 mg mL<sup>-1</sup>) and H<sub>3</sub>PO<sub>4</sub> (1%, v/v) by pipetting, and the mixture was then deposited onto the MALDI target for MS detection. The enrichment of peptides using the commercial Zip-TipC18 pipette tip was carried out according to the standard procedure provided by the technical note of MILLIPORE Corporation (http://www.millipore.com/techpublications/tech1/tn072).

#### Materials characterization

Scanning electron microscopy (SEM) images were obtained on a field emission scanning electron microscope (FESEM, S4800, Hitachi) equipped with an energy-dispersive X-ray spectrometer (EDX, JEOLJXA-840). Transmission electron microscopy (TEM) and high-resolution TEM (HRTEM) images were taken with a FEI Tecnai G2 S-Twin transmission electron microscope

operated at 200 kV. Powder X-ray diffraction (XRD) patterns were collected on a Bruker D8FOCUS X-ray diffractometer using Cu K $\alpha$  radiation ( $\lambda=1.5406~\mbox{Å}$ ) at a scanning rate of  $15^{\circ}$  min $^{-1}$  and a detection range from  $5^{\circ}$  to  $80^{\circ}$ . Fourier-transform infrared spectroscopy (FTIR) analyses were carried out on a Bruker Vertex 70 FTIR Spectrometer. Magnetization measurements were carried out with a superconducting quantum interface device (SQUID) magnetometer (Quantum Design MPMS XL) at 300 K. UV/visible absorption spectra and the adsorption curves were obtained using a Shimadzu UV-1750 UV/visible spectrophotometer.

MS analysis was carried out on a Bruker Autoflex III MALDITOF MS (Bruker Daltonics Bremen, Germany) equipped with a 355 nm Nd-YAG laser. The acceleration voltage and repetition rate were set at 19 kV and 200 Hz, respectively, whilst the laser power was optimized until the best-quality spectrum was obtained. For peptide mass fingerprinting (PMF) data, 600 laser shots were accumulated for each spectrum, which was obtained in positive ion reflection mode and analyzed by Bruker Daltonics flex Analysis software. Search parameters: fragment ion spectra were submitted to MASCOT (http://www.matrixscience.com/) for database search and identification of corresponding peptides employing the following search parameter settings. Database: SwissProt; enzyme: trypsin; maximum of missed cleavages: 1; taxonomy: mammalia (mammals); peptide tolerance: 1.2 Da; mass values: monoisotopic.

#### Results and discussion

As illustrated in Scheme 1, the synthetic strategy of GO–Fe<sub>3</sub>O<sub>4</sub> mainly includes functionalizing the as-synthesized GO with MPS to form GO–MPS, grafting Fe<sub>3</sub>O<sub>4</sub> particles onto the GO–MPS *via* a covalent linker to produce GO–Fe<sub>3</sub>O<sub>4</sub>, and reducing the GO–Fe<sub>3</sub>O<sub>4</sub> to rGO–Fe<sub>3</sub>O<sub>4</sub> in alkaline aqueous solution with urea as the pH adjustor. The MPS is a good linking reagent with bifunctional reactive groups. Its trimethoxysilane head group can be chemically bound to GO *via* hydroxyl and epoxy groups on it, whereas the mercapto group in the tail is able to form strong covalent bonds with Fe atoms of the Fe<sub>3</sub>O<sub>4</sub>. Therefore, the Fe<sub>3</sub>O<sub>4</sub> particles with different sizes and morphology can be easily covalently grafted onto the functional GO under mild reaction conditions.

The morphologies and structures of the as-prepared GO and the GO-Fe<sub>3</sub>O<sub>4</sub> composites were characterized using scanning electron microscopy (SEM) and transmission electron microscopy (TEM). Fig. 1 gives the representative SEM and TEM images of the GO (SEM: Fig. 1a; TEM: Fig. 1b), and the GO-Fe<sub>3</sub>O<sub>4</sub> composites (SEM: Fig. 1c and d; TEM: Fig. 1e-g). Compared with bare GO sheets (Fig. 1a and b), there are many Fe<sub>3</sub>O<sub>4</sub> particles attached on the surfaces of the treated GO sheets (Fig. 1c and d). The Fe<sub>3</sub>O<sub>4</sub> particles were well distributed on the GO sheets with nearly flat and several square micrometer area. Furthermore, no free or independent nanoparticles outside the GO sheets were observed, which indicates that all of the nanoparticles were attached to the GO. It can be distinctly observed from SEM images (Fig. 1d and S2†) that the Fe<sub>3</sub>O<sub>4</sub> particles of different sizes were immobilized on both sides of the GO, which will result in a better magnetic response. TEM images and EDX detection further provide the evidences of their assembly

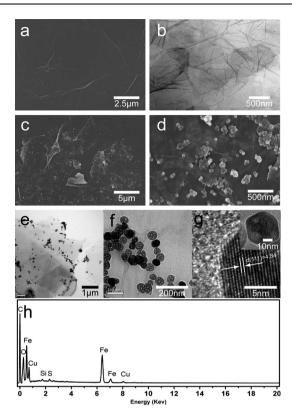


Fig. 1 SEM and TEM images of the GO (a and b) and the GO–Fe $_3$ O $_4$  (c–g), and EDX spectrum of the GO–Fe $_3$ O $_4$  (h).

(Fig. 1e–h). The average diameter of the attached  $Fe_3O_4$  nanoparticles is about 60 nm, as shown in Fig. 1f. According to the HRTEM image (Fig. 1g), the lattice fringe spacing (0.48 nm) agrees well with the lattice spacing of (111) planes of cubic magnetite. The EDX spectrum shown in Fig. 1h indicates that the  $GO-Fe_3O_4$  composites contain C, Fe, O, Si and S elements, while the Cu element arises from the sample carrier. It can be confirmed from these above results that even distribution of the  $Fe_3O_4$  nanoparticles on the whole GO nanosheets can be attributed to uniform distribution of the covalent bonding sites of -SH groups.

The grafting process can be verified by FTIR and UV-vis spectra. As shown by the FTIR in Fig. 2a, the presence of O-H  $(v_{(O-H)} \text{ at } 3412 \text{ cm}^{-1}), C=O(v_{(C=O)} \text{ at } 1733 \text{ cm}^{-1} \text{ in carbonyl}$ groups), C=C ( $v_{(C=C)}$  at 1621 cm<sup>-1</sup>), and C-O ( $v_{(C-OH)}$  at 1226 cm<sup>-1</sup> and  $\delta_{\rm (O-H)}$  at 1055 cm<sup>-1</sup>) indicates the successful preparation of GO, which is in accordance with previously reported results.1 After modification with MPS, the greatly increased intensity of the peaks at 2925 and 2854 cm<sup>-1</sup> ( $v_{\rm (C-H)}$  of -CH<sub>3</sub> and -CH<sub>2</sub>), the decreased intensity of the peaks at 3409 cm<sup>-1</sup> ( $v_{\text{(O-H)}}$ ) and 1055 cm<sup>-1</sup> ( $v_{\text{(C-O)}}$  of C-O-C), and the presence of new peaks at 1093 cm<sup>-1</sup> ( $v_{(Si-O)}$ ) of Si-O-C) and 1377 cm $^{-1}$  ( $\delta_{(C-H)}$  of S–CH $_2$ ) confirm that MPS was covalently grafted on the GO. After immobilization of Fe<sub>3</sub>O<sub>4</sub> particles, a new characteristic peak (565 cm<sup>-1</sup>) was observed, corresponding to the stretching vibration of the Fe-O bond. As expected, the characteristic peaks of hydrophilic groups (e.g. -COOH and -OH) can still be observed with high intensity, indicating the whole modification process did not lead to the reduction of the

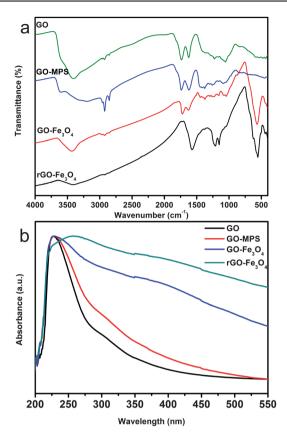


Fig. 2 FTIR spectra (a) and UV-vis spectra (b) of GO, GO–MPS, GO–Fe<sub>3</sub>O<sub>4</sub> and rGO–Fe<sub>3</sub>O<sub>4</sub>.

GO. Moreover, there exist more hydroxyl groups rooted in  $Fe_3O_4$ . Comparing with the spectrum of  $GO-Fe_3O_4$ , the disappearance of the peak at 1733 cm<sup>-1</sup> ( $v_{(C=O)}$  in carbonyl groups) in the rGO-Fe<sub>3</sub>O<sub>4</sub> FTIR provides obvious evidence of the deoxygenation of GO in alkaline solution, indicating the formation of rGO-Fe<sub>3</sub>O<sub>4</sub> composites, while the peaks at 1209 cm<sup>-1</sup> and 1153 cm<sup>-1</sup> should be ascribed to the formation of Si-O-Si bonds resulting from the further hydrolysis of partially reacted silane molecules in alkaline aqueous solution at 95 °C during the process of GO reduction.

The above conclusion can be further affirmed by the UV-vis absorption spectra (Fig. 2b). The characteristic peak at around 227 nm arising from the conjugated system of GO can be apparently observed in all the products. As expected, only slight peak shifting occurs after the covalent grafting of MPS and the immobilization of  $Fe_3O_4$  on the GO, indicating these modifications did not result in the unique GO structure being destroyed. However, after reduction, the rGO– $Fe_3O_4$  absorption peak red shifted to 256 nm because the loss of the most hydrophilic groups would enhance the electronic conjugation of graphene.  $^{33-35}$ 

The immobilization of the Fe<sub>3</sub>O<sub>4</sub> particles on GO can be further confirmed by XRD analysis. Fig. 3 shows the XRD patterns of Fe<sub>3</sub>O<sub>4</sub>, GO, GO–MPS, GO–Fe<sub>3</sub>O<sub>4</sub> and rGO–Fe<sub>3</sub>O<sub>4</sub> composites. The GO shows a very sharp diffraction peak at 10.3° corresponding to the (002) reflection of the GO. Compared with that of the GO, a broader and slightly red shifted diffraction peak is observed in the GO–MPS sample. The modification of the MPS could result in a subtle difference in the horizontal

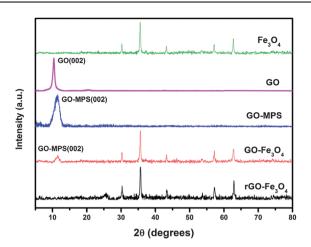


Fig. 3 XRD patterns of  $Fe_3O_4$ , GO, GO–MPS, GO– $Fe_3O_4$  and  $rGO-Fe_3O_4$ .

offset between layers in the GO–MPS.<sup>36,37</sup> After attachment of Fe<sub>3</sub>O<sub>4</sub> particles, some additional diffraction peaks appeared. They can be assigned to the characteristic diffraction peaks from Fe<sub>3</sub>O<sub>4</sub> (JCPDS card no. 65-3107), which is consistent with the precursor Fe<sub>3</sub>O<sub>4</sub> particles. The above results demonstrate that structures of the Fe<sub>3</sub>O<sub>4</sub> particles and the GO are almost unchanged after the mild modification. However, a visible shift of the (002) reflection peak from 10.3 to 25.3° on the rGO–Fe<sub>3</sub>O<sub>4</sub> XRD is observed, indicating the decrease of the interlayer space due to removing of the interbedded oxygen.<sup>37,38</sup>

Their magnetic property provides a rapid and convenient means for the separation of GO–Fe<sub>3</sub>O<sub>4</sub> and rGO–Fe<sub>3</sub>O<sub>4</sub> composites, being free of the repeated centrifugation for practical applications. The magnetic properties of the Fe<sub>3</sub>O<sub>4</sub> and GO–Fe<sub>3</sub>O<sub>4</sub> and rGO–Fe<sub>3</sub>O<sub>4</sub> were examined at 300 K *via* a superconducting quantum interface device (SQUID) magnetometer. As revealed in the hysteresis loops (Fig. 4), both the GO–Fe<sub>3</sub>O<sub>4</sub> and rGO–Fe<sub>3</sub>O<sub>4</sub> composites show a strong superparamagnetism at room temperature. The saturation magnetization values of Fe<sub>3</sub>O<sub>4</sub>, GO–Fe<sub>3</sub>O<sub>4</sub> and rGO–Fe<sub>3</sub>O<sub>4</sub> composites are 77.6, 33.0 and 36.9 emu g<sup>-1</sup>, respectively. Though the saturation magnetizations of GO–Fe<sub>3</sub>O<sub>4</sub> and rGO–Fe<sub>3</sub>O<sub>4</sub> composites are less than

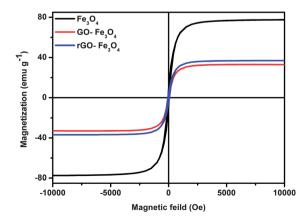


Fig. 4 Room-temperature hysteresis loops of  $Fe_3O_4$ ,  $GO-Fe_3O_4$  and  $rGO-Fe_3O_4$ .

that of the precursor magnetite particles, they are strong enough for effective magnetic separation.

The water-dispersibility of the GO/rGO-Fe<sub>3</sub>O<sub>4</sub> composites was investigated by dispersing them into water. For comparison, the Fe<sub>3</sub>O<sub>4</sub> particles and the graphene–Fe<sub>3</sub>O<sub>4</sub> composites prepared *via* the conventional solvothermal route (denoted as G-Fe<sub>3</sub>O<sub>4</sub>) were also investigated. As shown in Fig. S3,† an obvious aggregation and precipitation quickly occurred in the suspension of the graphene–Fe<sub>3</sub>O<sub>4</sub> composites prepared via the reported solvothermal method.38 However, the GO/rGO-Fe<sub>3</sub>O<sub>4</sub> suspensions were almost unchanged, indicating they are much more stable than the former. Furthermore, a comparison experiment was carried out to prove that the covalent conjugation is superior to the electrostatic attachment between Fe<sub>3</sub>O<sub>4</sub> and the GO or rGO. The GO/Fe<sub>3</sub>O<sub>4</sub> electrostatic attachment composite was first prepared by directly mixing the Fe<sub>3</sub>O<sub>4</sub> particles and the GO without modification with MPS under the same conditions as the GO-Fe<sub>3</sub>O<sub>4</sub>. Then the rGO/ Fe<sub>3</sub>O<sub>4</sub> electrostatically attached composite was obtained via the same reduction process as the rGO-Fe<sub>3</sub>O<sub>4</sub>. Subsequently, the GO/ Fe<sub>3</sub>O<sub>4</sub>, rGO/Fe<sub>3</sub>O<sub>4</sub>, GO-Fe<sub>3</sub>O<sub>4</sub> and rGO-Fe<sub>3</sub>O<sub>4</sub> were respectively dispersed into water by agitation on a shaker for 30 min. Then, these suspensions were attracted using the same magnet for 10 min. The experimental results indicate that, for the GO/Fe<sub>3</sub>O<sub>4</sub> and the rGO/Fe<sub>3</sub>O<sub>4</sub> systems, as revealed in Fig. 5a and b, although the Fe<sub>3</sub>O<sub>4</sub> particles attached to the GO or rGO can be attracted to one side of the vial near the magnet, the separated solution is still opaque due to the existence of suspended GO or rGO sheets. By contrast, the GO-Fe<sub>3</sub>O<sub>4</sub> and the rGO-Fe<sub>3</sub>O<sub>4</sub> composites can be easily separated by the external magnet (see Fig. 5c and d), indicating that the GO-Fe<sub>3</sub>O<sub>4</sub> and the rGO-Fe<sub>3</sub>O<sub>4</sub> composites with good water dispersibility and excellent magnetic response are in favour of absorption and convenient separation of biomolecules. Moreover, the dispersed systems with good water dispersibility are stable enough to meet our testing requirement.

Affinity immobilization and separation of proteins and peptides have aroused great interest in the past two decades, which are widely applied in biosensors, biocatalysis, and purification of biosamples. Since GO with many hydrophilic functional groups can incorporate biomolecules by covalent, electrostatic or hydrogen bonding interactions, Az, at the GO–Fe<sub>3</sub>O<sub>4</sub> magnetic composites can be applied to immobilize and magnetically separate proteins from solution. As an example, the

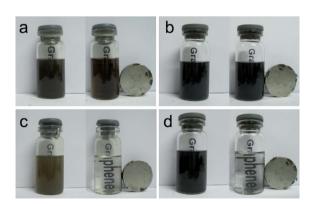
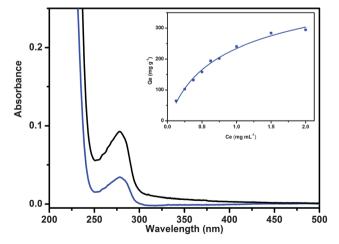


Fig. 5 Magnetic response photos of the  $GO/Fe_3O_4$  (a), the  $rGO/Fe_3O_4$  (b), the  $GO-Fe_3O_4$  (c) and the  $rGO-Fe_3O_4$  (d) aqueous solution systems.

GO-Fe<sub>3</sub>O<sub>4</sub> (60 nm) composites have been investigated. Bovine serum albumin (BSA), a very abundant protein in bovine blood, was chosen as a model protein to evaluate the ability of GO-Fe<sub>3</sub>O<sub>4</sub> composites for protein immobilization. The amount of Fe<sub>3</sub>O<sub>4</sub> particles attached on the surface of GO is a critical factor in determining the adsorption of proteins. A higher loading of Fe<sub>3</sub>O<sub>4</sub> will enhance the magnetic properties of GO-Fe<sub>3</sub>O<sub>4</sub>,44 while it will decrease the exposed surface area of GO simultaneously, thus resulting in a lower adsorption capacity. Therefore, three ratios of GO to Fe<sub>3</sub>O<sub>4</sub> composites have been prepared for investigating the protein immobilization. As shown in Fig. S4,† the adsorption ability increase with the ratio of GO to Fe<sub>3</sub>O<sub>4</sub>. However, it is hard to magnetically separate the GO-Fe<sub>3</sub>O<sub>4</sub> composites rapidly when the ratio of GO to Fe<sub>3</sub>O<sub>4</sub> is as high as 2: 1, due to the low Fe<sub>3</sub>O<sub>4</sub> content in the composites. When the ratio of Fe<sub>3</sub>O<sub>4</sub> to GO is 1:1, a relatively optimal result can be obtained. Therefore, we chose GO-Fe<sub>3</sub>O<sub>4</sub> composites with the ratio of Fe<sub>3</sub>O<sub>4</sub> to GO (1:1) as models to further investigate their applications. Meanwhile, adsorption abilities of Fe<sub>3</sub>O<sub>4</sub>, GO-Fe<sub>3</sub>O<sub>4</sub> and rGO-Fe<sub>3</sub>O<sub>4</sub> were compared using BSA solution (1 mg mL<sup>-1</sup>). Apparently, the composites show much better adsorption abilities than the bare Fe<sub>3</sub>O<sub>4</sub> particles, and the GO-Fe<sub>3</sub>O<sub>4</sub> composites show the best adsorption ability (Fig. S5†). This can be attributed to the high surface area and the hydrophilic functional groups on the surface of GO in the GO-Fe<sub>3</sub>O<sub>4</sub> composites.

Fig. 6 shows the UV-vis absorption spectra of the BSA solution with initial concentration of  $0.125~\text{mg}~\text{mL}^{-1}$  before and after adsorption by the GO–Fe<sub>3</sub>O<sub>4</sub> composites. The wavelength with maximum absorbance (280 nm) was used as detecting wavelength. For  $0.125~\text{mg}~\text{mL}^{-1}$  of the BSA solution, 51.65% of the BSA can be removed or immobilized. The adsorption capacity can be judged from the absorption curve (inset). Obviously, the adsorbed amount of the BSA gradually increased with the increase of the equilibrium concentration. When the concentration of BSA is  $2.0~\text{mg}~\text{mL}^{-1}$ , the adsorption capacity of the proteins adsorbed by GO–Fe<sub>3</sub>O<sub>4</sub> composites is  $294.54~\text{mg}~\text{g}^{-1}$ . The high adsorption capacity stems from strong interactions



**Fig. 6** UV-vis absorption spectra of  $0.125~\text{mg}~\text{mL}^{-1}$  of the BSA solution before (black) and after adsorption by the GO–Fe<sub>3</sub>O<sub>4</sub> (blue) and adsorption isotherm of different concentrations of BSA onto the GO–Fe<sub>3</sub>O<sub>4</sub> composite (inset).

between the BSA protein and the oxygen-containing functional groups on the GO-Fe<sub>3</sub>O<sub>4</sub> composites.

In proteomics, many low-abundance endogenous peptides can be used as biomarkers with high clinical sensitivity and specificity, which could provide valuable information for the detection of many diseases and elucidation of pathological variations. 45 However, it is a great challenge to detect the low-abundance peptides directly due to their extremely low concentrations and the interference of salts. Graphene has a great potential for adsorption of biomolecules due to its high specific surface area, hydrophobicity and unique conjugation structure. Thus, the rGO-Fe<sub>3</sub>O<sub>4</sub> composites can be applied as an affinity probe for enrichment of low-abundance peptides by taking advantage of the special properties of rGO and the rapid magnetic response of Fe<sub>3</sub>O<sub>4</sub>. As an example, the tryptic digests of BSA are employed to demonstrate the possibility for enrichment and rapid separation of low-abundance peptides. Fig. 7 shows the MALDI-TOF mass spectra of the diluted tryptic digest of BSA before and after treatment with the rGO-Fe<sub>3</sub>O<sub>4</sub> composites. Apparently, only three peptides (marked with '\*') with the low signal-to-noise (S/N) ratio can be detected from the diluted BSA digest (Fig. 7a) due to the low concentration of target peptides and the interference of salts used in the digestion process. However, after enrichment with the rGO-Fe<sub>3</sub>O<sub>4</sub> composites, nineteen peptides with the sequence coverage of 21% (Table S1†) could be assigned to BSA with increased S/N ratio and intensity, indicating the rGO-Fe<sub>3</sub>O<sub>4</sub> composites can enrich the low-abundance peptides (Fig. 7b). For comparison, the pure Fe<sub>3</sub>O<sub>4</sub> particles are used to enrich the peptides. After treatment using the Fe<sub>3</sub>O<sub>4</sub> particles, nine peptides can be observed with slightly increased S/N ratio, while most of the peptides with low molecular weight are lost (Fig. S7a†). In addition, a commercial Zip-TipC18 pipette tip is also used to enrich the peptides. Apparently, only a few peptides with lower S/N ratio can be observed (Fig. S7b†) after the solution was treated by a Zip-TipC18 pipette tip. The above results indicate the prepared rGO-Fe<sub>3</sub>O<sub>4</sub> composites show good potential for enriching low molecular weight peptides. The hydrophobic interactions between the rGO-Fe<sub>3</sub>O<sub>4</sub> composites and the exposed hydrophobic residues of the low molecular

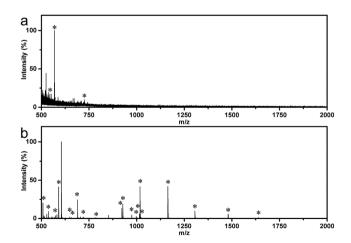


Fig. 7 MALDI-TOF mass spectra of the diluted digest of BSA (5 fmol  $\mu L^{-1}$ ) without any treatment (a), and after enrichment with rGO–Fe<sub>3</sub>O<sub>4</sub> composites (b).

weight peptides should be responsible for the adsorption of the low molecular weight peptides.

#### Conclusion

In conclusion, the GO–Fe<sub>3</sub>O<sub>4</sub> and the rGO–Fe<sub>3</sub>O<sub>4</sub> composites with good water dispersibility, high affinity and rapid magnetic response have been prepared. Significantly, the GO–Fe<sub>3</sub>O<sub>4</sub> composites with hydrophilic groups can be used as an effective absorbent or carriers for rapid separation and immobilization of proteins, while the water dispersible rGO–Fe<sub>3</sub>O<sub>4</sub> composites can be used for enrichment of peptides by virtue of their high surface area and special structure. Furthermore, the synthetic route can also be expanded to graphene oxide or graphene systems with other nanostructures. The GO/rGO–Fe<sub>3</sub>O<sub>4</sub> composites could also be applied as a carrier for enzyme immobilization for specific functions, capture of biomolecules for water purification, and enrichment of small biomolecules for biomedical applications.

#### Acknowledgements

This work was supported by the National Natural Science Foundation of China (NNSFC) (Grant no. 20871083 and 21171161).

#### Notes and references

- H. Wang, Q. Zhang, X. Chu, T. Chen, J. Ge and R. Yu, Angew. Chem., Int. Ed., 2011, 50, 7065–7069.
- 2 S. Stankovich, D. A. Dikin, G. H. Dommett, K. M. Kohlhaas, E. J. Zimney, E. A. Stach, R. D. Piner, S. T. Nguyen and R. S. Ruoff, *Nature*, 2006, 442, 282–286.
- 3 V. C. Tung, M. J. Allen, Y. Yang and R. B. Kaner, *Nat. Nanotechnol.*, 2009, 4, 25–29.
- 4 R. S. Ruoff, J. Mater. Chem., 2011, 21, 3272.
- 5 H.-J. Song, X.-H. Jia, N. Li, X.-F. Yang and H. Tang, J. Mater. Chem., 2012, 22, 895–902.
- 6 V. Chandra, J. Park, Y. Chun, J. W. Lee, I.-C. Hwang and K. S. Kim, ACS Nano, 2010, 4, 3979–3986.
- 7 S. Myung, A. Solanki, C. Kim, J. Park, K. S. Kim and K.-B. Lee, Adv. Mater., 2011, 23, 2221–2225.
- 8 S. Guo and S. Dong, J. Mater. Chem., 2011, 21, 18503-18516.
- 9 W. Shi, J. Zhu, D. H. Sim, Y. Y. Tay, Z. Lu, X. Zhang, Y. Sharma, M. Srinivasan, H. Zhang, H. H. Hng and Q. Yan, *J. Mater. Chem.*, 2011, 21, 3422–3427.
- 10 Y. Lin, Y. Tao, F. Pu, J. Ren and X. Qu, Adv. Funct. Mater., 2011, 21, 4565–4572.
- 11 L. Feng, L. Wu, J. Wang, J. Ren, D. Miyoshi, N. Sugimoto and X. Qu, Adv. Mater., 2012, 24, 125–131.
- 12 B. Nam, H.-J. Lee, H. Goh, Y. B. Lee and W. S. Choi, J. Mater. Chem., 2012, 22, 3148–3153.
- 13 Y. Zhang, J. Zhang, X. Huang, X. Zhou, H. Wu and S. Guo, *Small*, 2012, **8**, 154–159.

- 14 L. A. L. Tang, J. Wang and K. P. Loh, J. Am. Chem. Soc., 2010, 132, 10976–10977.
- 15 B. Gulbakan, E. Yasun, M. I. Shukoor, Z. Zhu, M. You, X. Tan, H. Sanchez, D. H. Powell, H. Dai and W. Tan, J. Am. Chem. Soc., 2010, 132, 17408–17410.
- 16 Y. Deng, D. Qi, C. Deng, X. Zhang and D. Zhao, J. Am. Chem. Soc., 2008, 130, 28–29.
- 17 H. Chen, C. Deng and X. Zhang, Angew. Chem., 2010, 122, 617-621.
- 18 Y. Li, C. Dong, J. Chu, J. Qi and X. Li, Nanoscale, 2011, 3, 280-287.
- 19 G. Cheng, J. L. Zhang, Y. L. Liu, D. H. Sun and J. Z. Ni, Chem. Commun., 2011, 47, 5732–5734.
- B. Li, H. Cao, J. Shao and M. Qu, Chem. Commun., 2011, 47, 10374-10376.
- 21 B. Li, H. Cao, J. Shao, M. Qu and J. H. Warner, J. Mater. Chem., 2011. 21, 5069–5075.
- 22 C. Deng, J. Meng and C. Shi, Chem. Commun., 2012, 48, 2418–2420.
- 23 X. Li, X. Huang, D. Liu, X. Wang, S. Song, L. Zhou and H. Zhang, J. Phys. Chem. C, 2011, 115, 21567–21573.
- 24 H. He and C. Gao, ACS Appl. Mater. Interfaces, 2010, 2, 3201-3210.
- 25 X. Yang, X. Zhang, Y. Ma, Y. Huang, Y. Wang and Y. Chen, J. Mater. Chem., 2009, 19, 2710–2714.
- 26 H. Sun, L. Cao and L. Lu, Nano Res., 2011, 4, 550-562.
- 27 X. Huang, X. Qi, F. Boey and H. Zhang, Chem. Soc. Rev., 2012, 41, 666–686.
- 28 W. S. Hummers and R. E. Offeman, J. Am. Chem. Soc., 1958, 80, 1339.
- 29 D. C. Marcano, D. V. Kosynkin, J. M. Berlin, A. Sinitskii, Z. Sun, A. Slesarev, L. B. Alemany, W. Lu and J. M. Tour, ACS Nano, 2010, 4, 4806–4814.
- 30 P. Lu, J.-L. Zhang, Y.-L. Liu, D.-H. Sun, G.-X. Liu, G.-Y. Hong and J.-Z. Ni, *Talanta*, 2010, 82, 450–457.
- 31 X. Fan, W. Peng, Y. Li, X. Li, S. Wang, G. Zhang and F. Zhang, Adv. Mater., 2008, 20, 4490–4493.
- 32 Z. Lei, L. Lu and X. S. Zhao, Energy Environ. Sci., 2012, 5, 6391–6399.
- 33 Y. Liang, D. Wu, X. Feng and K. Müllen, Adv. Mater., 2009, 21, 1679–1683.
- 34 S. Villar-Rodil, J. I. Paredes, A. Martinez-Alonso and J. M. D. Tascon, *J. Mater. Chem.*, 2009, 19, 3591–3593.
- 35 B.-S. Kong, J. Geng and H.-T. Jung, Chem. Commun., 2009, 2174–2176.
- 36 B. Lukose, A. Kuc and T. Heine, Chem.-Eur. J., 2011, 17, 2388-2392.
- 37 G. Xie, P. Xi, H. Liu, F. Chen, L. Huang, Y. Shi, F. Hou, Z. Zeng, C. Shao and J. Wang, *J. Mater. Chem.*, 2012, **22**, 1033–1039.
- 38 K. Zhou, Y. Zhu, X. Yang and C. Li, New J. Chem., 2010, 34, 2950–
- 39 S. Y. Lee, S. Lee, I. H. Kho, J. H. Lee, J. H. Kim and J. H. Chang, Chem. Commun., 2011, 47, 9989–9991.
- 40 J. Lee, H. B. Na, B. C. Kim, J. H. Lee, B. Lee, J. H. Kwak, Y. Hwang, J.-G. Park, M. B. Gu, J. Kim, J. Joo, C.-H. Shin, J. W. Grate, T. Hyeon and J. Kim, J. Mater. Chem., 2009, 19, 7864–7870.
- 41 M. Zhang, X. He, L. Chen and Y. Zhang, *Nanotechnology*, 2011, 22, 065705.
- 42 D. R. Dreyer, S. Park, C. W. Bielawski and R. S. Ruoff, *Chem. Soc. Rev.*, 2010, 39, 228–240.
- 43 S. Park, J. An, R. D. Piner, I. Jung, D. Yang, A. Velamakanni, S. T. Nguyen and R. S. Ruoff, *Chem. Mater.*, 2008, 20, 6592–6594.
- 44 H.-P. Cong, J.-J. He, Y. Lu and S.-H. Yu, Small, 2010, 6, 169–173.
- 45 M. Swami, Nat. Rev. Cancer, 2010, 10, 597.