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Convenient divergent synthesis of linear-dendron block polyphosphoesters via acyclic diene metathesis polymerization†

Liang Ding,*a Jun Qiu,a Jun Weia and Zhenshu Zhu*bc

A novel divergent approach was developed for the first time for the synthesis of linear-dendritic polyphosphoester (PPE) structures using an acrylated poly(ethylene glycol) methyl ether as a linear macromolecular chain stopper by acyclic diene metathesis (ADMET) polymerization of phosphoester functional asymmetric α, ω -diene monomers. This synthesis is remarkable, because unlike all others, each low-generation linear-dendron copolymer could be readily converted, by thiol-Michael addition click reaction and esterification, to a new selective macromolecular chain stopper by subsequent ADMET polymerization to synthesize high-generation linear-dendron block PPEs, and it requires no means of purification other than a simple precipitation. The prepared linear-dendritic PPEs can self-assemble spontaneously in a selective solvent to form polymeric nanoparticles, which were characterized in detail by DLS, AFM, and TEM analyses. To the best of our knowledge, this is the first report that describes the synthesis of linear-dendron-like block PPEs via ADMET polymerization. Consequently, this provides a versatile strategy not only for the synthesis of biodegradable and amphiphilic block PPEs with lineardendron-like architecture but also for the fabrication of biocompatible nanoparticles with a suitable size for biomedical applications.

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Introduction

The synthesis of complex, branched macromolecules with welldefined polymer architecture represents an intriguing current challenge in polymer chemistry.1 Linear-dendritic block copolymers, in which a dendritic segment is attached to a linear polymer chain, are interesting with respect to both chemical and physical behavior resulting from the constraints imposed upon the branched block by the topology.2-4 With the different molecular structures and functions, the linear-dendritic block copolymers can be self-assembled into various complex aggregates, which can be used for drug and gene delivery. 5-8 Usually, the perfectly linear-branched dendrons and linear-hyperbranched materials exhibit the linear-dendritic structure. 9,10 Though these two macromolecules share many architectural similarities such as multiple terminal groups and globular shape, they differ in their methods of preparation.¹⁰ The synthesis of several linear-hyperbranched block copolymers has been reported, being conducted in fewer synthetic steps and thus on a large scale.11 In contrast, only a few papers have detailed the preparation of block copolymers with perfectly branched dendrimer blocks that can be obtained in multi-step divergent processes recently. 8,12,13 Most of them are synthesized through three synthetic strategies: coupling of dendrons with a linear segment; the divergent construction of a dendron from the end group of a linear polymer chain; and the growth of a linear block from the apex of a dendron. However, utilizing the common polymerization methods for synthesis of lineardendritic block copolymers directly is quite rare.2b

Olefin metathesis based step-growth polymerization, termed acyclic diene metathesis (ADMET) polymerization, is performed with α, ω -dienes, and proceeds by release of ethylene as a condensate.14 This polymerization technique is usually applied to prepare polymers with a wide composition range through monomer design.15 Importantly, Meier has broadened the applications of this technique and introduced the concept of a selective head-to-tail ADMET polymerization using a monomer containing both a terminal double bond and an acrylate. Such monomers polymerize with high cross metathesis selectivity, opening access to different polymer architectures, if a selective and irreversible chain transfer agent (mono- or multi-functional acrylate) is added.16 The chain transfer agent allows controlling

aSchool of Materials Engineering, Yancheng Institute of Technology, Yancheng, 224051, China. E-mail: dl1984911@ycit.edu.cn; Fax: +86 515 88298872; Tel: +86 515 88232258

^bKey Laboratory of Drug Quality Control and Pharmacovigilance (China Pharmaceutical University), Ministry of Education, Department of Pharmaceutical Analysis, China Pharmaceutical University, Nanjing 210009, China. E-mail: zhuzs@ cpu.edu.cn

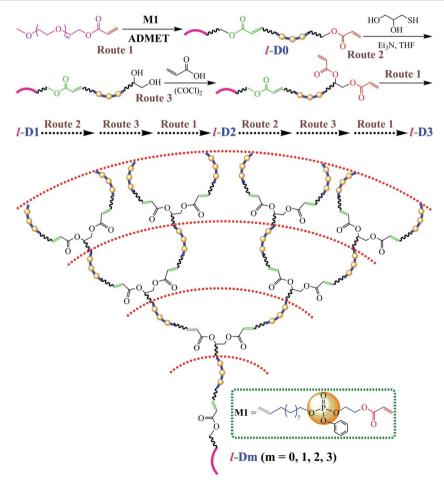
Department of Chemical and Biomolecular Engineering, National University of Singapore, 117578, Singapore

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the molecular weight and direct functionalization of the ADMET polymer by reacting selectively with one of the end groups (terminal double bond). By employing this approach, a number of issues in conventional ADMET polymerization were avoided. In a typical ADMET polymerization the internal alkenes are still metathesis-active, which results in interchange reactions. The presence of such reactions would have eliminated the dendritic architecture, but this was avoided by virtue of the fact that the internal acrylates are too electron deficient to undergo further reaction with the metathesis catalyst. 17 Besides, the driving force of ADMET polymerization is the removal of ethylene, which prevents the otherwise thermodynamically mandated depolymerization by ethenolysis. However, when a relatively electron-rich olefin is coupled with one that is relatively electron-deficient, the reaction is irreversible under typical metathesis conditions.18 Consequently, the polymer is not susceptible to depolymerization caused by the presence of ethylene. This allows the high vacuum conditions, a requisite in standard ADMET polymerization, to be eliminated.19 As a result, this approach would lead to a wide range of unsaturated and saturated polymers bearing different functionalities with the possibility to modify the polyolefin architectures by placing precise functional groups throughout the backbone.20 Nevertheless, to the best of our knowledge, to date ADMET

polymerization has not been applied to the synthesis of perfect dendrons, especially linear-dendron block copolymers.²¹ Therefore, the development of functional linear-dendron metathesis polyolefins bearing reactive substituents is of significant importance.

In recent projects, we have been interested in the development of novel potentially biodegradable and biocompatible polyphosphoesters (PPEs). PPEs can be degraded by several enzymes such as phosphatases and phosphodiestereases and/ or by basic or acidic hydrolysis.22 In spite of this obvious benefit, PPEs are only scarcely found in recent studies, even if they are easily accessible and allow the feasible synthesis of a great variety of functional materials. The combination of phosphorus chemistry with metathesis allows tailoring of the polymer functionality due to the high functional group tolerance of modern ruthenium metathesis catalysts.23 Wurm recently presented an expansion of the metathesis polymerization towards PPEs from a step-growth ADMET polymerization²⁴ to the chaingrowth ROMP.25 Based on these precedents, we recently reported the synthesis of diverse metathesis PPE architectures, such as linear (alternating and block), star-shaped, cyclic, and hyperbranched PPEs via ADMET polymerization.26 However, more complex linear-dendron block PPEs have not been synthesized and further exploited.



Scheme 1 Illustration for preparation of linear-dendron block polyphosphoesters via ADMET polymerization.

Toward this attractive goal, herein, we present a divergent synthesis of novel linear-dendron block PPEs via ADMET polymerization, based on readily available PPE as the repeating unit. A phosphoester functional asymmetric α,ω-diene monomer containing both a terminal olefin and an acrylate group would polymerize only by head-to-tail addition and could exhibit the highest terminal olefin-to-acrylate metathesis selectivity. We introduced for the first time a macromolecule containing a terminal acrylate group to this polymerization as a selective chain stopper, resulting in the formation of a linear diblock PPE (I-D0). Then, the acrylate end group, which is directly obtained with very high end group fidelity, would be readily converted to multifunctional acrylate end groups by thiol-Michael addition click reaction and esterification. These multifunctional acrylates could act as new selective macromolecular chain stoppers by subsequent ADMET polymerization reactions to prepare linear-dendron (**l-Dm**, m = 1, 2, 3) block PPEs (Scheme 1). In all cases, the molecular weight was successfully controlled by choosing the chain stopper/monomer ratio. This novel route is noteworthy due to its ability to rapidly produce high-generation PPE dendrimers of high purity with no means of purification other than a simple precipitation.

Experimental section

Materials

Paper

3-Mercapto-1,2-propanediol (90+%), oxalyl chloride [(COCl)₂, [1,3-bis(2,4,6-trimethylphenyl)-2-imidazolidinylidene]dichloro(o-isopropoxyphenylmethylene)-ruthenium (Hoveyda-Grubbs second generation catalyst) (98%), acrylic acid (99%), and methoxypolyethylene glycols (average $M_{\rm n}=1900$) were purchased from Energy Chemical and used as received without purification. Solvents were distilled over drying agents under nitrogen prior to use. Triethylamine (Et₃N) and pyridine were freshly distilled and dried. The phosphoester functional asymmetric α, ω -diene monomer (M1) and poly(ethylene glycol) methyl ether acrylate were synthesized as reported previously by our group.26c

Characterization

FT-IR spectra were recorded on a Bruker Tensor 37 in the region of 4000–400 cm^{-1} using KBr pellets. 1 H (500 MHz) and 13 C (125 MHz) NMR spectra were recorded using tetramethylsilane as an internal standard in CDCl3 on a Bruker DPX spectrometer. Phosphoric acid (85%) was used as the external reference for ³¹P NMR analysis. Multiangle laser light scattering-gel permeation chromatography (MALLS-GPC) was employed to evaluate the molecular weights of the synthesized samples. The MALLS-GPC system consisted of a Waters 2690D Alliance liquid chromatography system, a Wyatt Optilab DSP differential refractometer detector (690 nm, Wyatt Technology), and a Wyatt DAWN EOS MALLS detector (30 mW GaAs linearly polarized laser, 690 nm). Two chromatographic columns (Styragel HR3, HR4) with a precolumn were used in series. THF was used as the mobile phase at a flow rate of 1.0 mL min⁻¹ at 30 °C. A 100 μL sample of a 3.0 mg mL⁻¹ solution, which was filtered through a 0.2 μm

Whattman filter prior to use, was injected for all measurements. The refractive index increment (dn/dc) of polymer samples in THF was determined with a Wyatt Optilab DSP differential refractometer at 690 nm. Data analysis was performed with Astra software (ver. 4.90.04, Wyatt Technology). The hydrodynamic diameter was determined by means of dynamic light scattering (DLS) analysis using a Malvern Zetasizer Nano-ZS light scattering apparatus (Malvern Instruments, U.K.) with a He-Ne laser (633 nm, 4 mW). Atomic force microscopy (AFM) observations were performed on a SPM AJ-III atomic force microscope at a measure rate of 1.0005 Hz in the tapping mode, and the AFM images were obtained at room temperature in air. Transmission electron microscopy (TEM) was performed on a JEM-200CX microscope operating at an acceleration voltage of 200 kV.

Polymerization reactions were carried out in Schlenk tubes under a dry nitrogen atmosphere.

Synthesis of linear-dendron block PPE terminated with the acrylate group (l-D0-Acrylate) via ADMET polymerization (route 1)

In a nitrogen-filled Schlenk tube, phosphoester functional asymmetric α, ω -diene M1 and a selective chain stopper (poly(ethylene glycol) methyl ether acrylate) in a 5:1 molar ratio were degassed by three freeze-vacuum-thaw cycles. The mixture was heated to 40 °C while stirring and then a solution of 0.5 mol% Hoveyda-Grubbs second generation catalyst (to a single acrylate group) in 0.5 mL of CHCl₃ degassed with the same procedure was added. After the reaction mixture was stirred for 24 h, the polymerization was quenched by adding THF (2 mL) and ethyl vinyl ether with stirring for 30 min. The solution was precipitated into an excess of methanol, and the precipitate was isolated by filtration and dried under vacuum to give **l-D0-Acrylate** block PPE. ¹H NMR (CDCl₃), δ (ppm): 7.52-7.43 (m, m-ArH), 7.21-7.11 (m, o-ArH + p-ArH), 7.02-6.90(m, OCOCH=CH), 6.42-6.35 (d, OCOCH=CH₂), 6.23-6.17 (m, OCOCH=CH₂) $OCOCH = CH_2$), 5.88-5.79 (d, $OCOCH = CH + CH_2 = CHCH_2O$), 4.39-4.16 (m, $CH_2CH_2OCOCH=CH_2 + CH_2=CHCOOCH_2CH_2$), 3.93-3.71 (m, PEG backbone + CH_2OPOCH_2), 3.24-3.15 (m, OCH_3), 1.93-1.84 (m, $CH_2 = CHCH_2$, 1.65-1.47 OCH₂CH₂CH₂), 1.39-1.06 (m, CH₂=CHCH₂(CH₂)₆). ³¹P NMR, δ (ppm): -4.20. MALLS-GPC: $M_n = 4100$, $M_w/M_n = 1.31$; ¹H NMR: $M_{\rm n} = 3900$.

End acrylate group functionalization of l-D0-Acrylate block PPE via a thiol-Michael addition click reaction (route 2)

The ADMET copolymer I-D0-Acrylate block PPE (1.0 mmol), 3mercapto-1,2-propanediol (0.22 g, 2 mmol) and triethylamine (0.2 g, 2 mmol) were dissolved in 10 mL of THF in a Schlenk tube and stirred overnight at room temperature. The product was then precipitated quantitatively from methanol and dried under vacuum for 24 h to give **l-D0** block PPE with two hydroxyl end groups (**l-D0–2OH**). ¹H NMR (CDCl₃), δ (ppm): 7.53–7.37 (m, m-ArH), 7.22–6.92 (m, o-ArH + p-ArH + OCOCH=CH), 5.91–5.85 (d, OCOCH=CH), 4.43-4.12 (m, $CH_2CH_2OCOCH=CH_2+CH_2=$ $CHCOOCH_2CH_2$), 4.03–3.70 (m, PEG backbone + CH_2OPOCH_2 +

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HOC H_2 CHOH), 3.23–3.11 (m, OC H_3), 2.93–2.81 (m, CH $_2$ SCH $_2$ CH $_2$), 2.61–2.39 (m, CH $_2$ SCH $_2$ CH $_2$), 1.96–1.72 (m, CH $_2$ =CHC H_2), 1.64–1.50 (m, OCH $_2$ CH $_2$ CH $_2$), 1.46–1.09 (m, CH $_2$ =CHCH $_2$ (CH $_2$) $_6$). ³¹P NMR, δ (ppm): –2.60. MALLS-GPC: $M_n = 4200$, $M_w/M_n = 1.30$; ¹H NMR: $M_n = 4100$.

Preparation of l-D0 block PPE bearing two acrylate end groups (l-D0-2Acrylate) (route 3)

Under a nitrogen atmosphere, (COCl)₂ (2.6 mL, 30 mmol) was added using a syringe to acrylic acid (0.43 g, 6 mmol), at room temperature with rapid stirring. After 6 h, the excess (COCl)₂ was removed in vacuo to yield acryloyl chloride, which was then added using a syringe to the solution of I-D0-2OH block PPE (0.6 mmol) in 10 mL of CH₂Cl₂ and dry triethylamine (1.0 mL, 7.5 mmol) at 0 °C. The reaction mixture was then allowed to warm to room temperature and stirred overnight. The precipitate was filtered off and the filtrate was washed with water; then the organic layers were dried over anhydrous Na₂SO₄, and the concentrated residue was precipitated twice from methanol and dried in a vacuum oven to afford two acrylate end-functionalized **l-D0** block PPE. ¹H NMR (CDCl₃), δ (ppm): 7.58–7.41 (m, m-ArH), 7.22–7.01 (m, o-ArH + p-ArH), 7.00–6.91 (m, OCOCH=CH), 6.39-6.32 (d, OCOCH= CH_2), 6.18-6.09 (m, OCOCH= CH_2), 5.83-5.72 (d, OCOCH=CH + CH₂=CHCH₂O), 4.82-4.79 (m, CH_2CHCH_2 , 4.47-4.21 (m, $CH_2OCOCH=CH_2 + CH_2=$ $CHCOOCH_2 + CH_2CHCH_2$, 4.00-3.73 (m, PEG backbone + CH_2OPOCH_2), 3.28–3.12 (m, OCH_3), 2.87 - 2.79 $CH_2SCH_2CH_2$), 2.66-2.35 (m, $CH_2SCH_2CH_2$), 2.03-1.82 (m, $CH_2 = CHCH_2$, 1.55-1.34 (m, $OCH_2CH_2CH_2$), 1.31-1.01 (m, CH₂=CHCH₂(CH₂)₆). ³¹P NMR, δ (ppm): -2.90. MALLS-GPC: $M_{\rm n} = 4400$, $M_{\rm w}/M_{\rm n} = 1.32$; ¹H NMR: $M_{\rm n} = 4300$.

Synthesis of linear-dendron block PPE terminated with a two acrylate group (l-D1-2Acrylate)

l-D1-2Acrylate block PPE was synthesized from l-D0-2Acrylate (0.1 mmol) using the same method as successive thiol-Michael addition click reaction of the acrylate end group, esterification of obtained two hydroxyl end groups, and ADMET polymerization with phosphoester functional asymmetric α, ω -diene M1 in a 1:10 molar ratio in the presence of 1 mol% (to total acrylate groups) Hoveyda–Grubbs second generation catalyst. ¹H NMR $(CDCl_3)$, δ (ppm): 7.53–7.37 (m, *m*-Ar*H*), 7.20–6.89 (m, *o*-Ar*H* + *p*-ArH + OCOCH = CH), 6.38-6.33 (d, OCOCH = CH_2), 6.12-6.04 (m, OCOCH=CH₂), 5.78-5.67 (d, OCOCH=CH + CH₂= CHCH₂O), 4.80-4.76 (m, CH₂CHCH₂), 4.49-4.36 (m, CH₂CH₂O- $COCH=CH_2 + CH_2=CHCOOCH_2CH_2 + CH_2CHCH_2$, 3.98-3.69 (m, PEG backbone + CH_2OPOCH_2), 3.34-3.16 (m, OCH_3), 2.91-2.82 (m, CH₂SCH₂CH₂), 2.68-2.39 (m, CH₂SCH₂CH₂), 2.01-1.78 CH₂=CHCH₂(CH₂)₆). ³¹P NMR, δ (ppm): -1.70. MALLS-GPC: $M_{\rm n} = 8500$, $M_{\rm w}/M_{\rm n} = 1.36$; ¹H NMR: $M_{\rm n} = 8300$.

Synthesis of linear-dendron block PPE bearing four acrylate end groups (l-D2-4Acrylate)

l-D2–4Acrylate block PPE was synthesized from **l-D1–2Acrylate** (0.1 mmol) using the same method as successive thiol-Michael

addition click reaction of two acrylate end groups, esterification of obtained four hydroxyl end groups, and ADMET polymerization with phosphoester functional asymmetric α, ω -diene M1 in a 1:20 molar ratio in the presence of 2 mol% (to total acrylate groups) Hoveyda-Grubbs second generation catalyst. ¹H NMR (CDCl₃), δ (ppm): ¹H NMR (CDCl₃), δ (ppm): 7.53–7.37 (m, m-ArH), 7.20–6.89 (m, o-ArH + p-ArH + OCOCH=CH), 6.38– 6.33 (d, OCOCH= CH_2), 6.12-6.04 (m, OCOCH= CH_2), 5.78-5.67 (d, OCOCH=CH + CH₂=CHCH₂O), 4.80-4.76 (m, CH₂CHCH₂), 4.49-4.36 (m, $CH_2CH_2OCOCH=CH_2+CH_2=CHCOOCH_2CH_2+$ CH_2CHCH_2), 3.98–3.69 (m, PEG backbone + CH_2OPOCH_2), 3.34– 3.16 (m, OCH_3), 2.91-2.82 (m, $CH_2SCH_2CH_2$), 2.68-2.39 (m, $CH_2SCH_2CH_2$), 2.01-1.78 (m, $CH_2=CHCH_2$), 1.53-1.37 (m, OCH₂CH₂CH₂), 1.35-1.03 (m, CH₂=CHCH₂(CH₂)₆). 31 P NMR, δ (ppm): -0.30. MALLS-GPC: $M_{\rm p} = 17 \ 100$, $M_{\rm w}/M_{\rm p} = 1.37$; ¹H NMR: $M_{\rm n} = 16~800$.

Synthesis of linear-dendron block PPE terminated with eight acrylate groups (I-D3-8Acrylate)

1-D3-8Acrylate block PPE was synthesized from 1-D2-4Acrylate (0.1 mmol) using the same method as successive thiol-Michael addition click reaction of four acrylate end groups, esterification of obtained eight hydroxyl end groups, and ADMET polymerization with phosphoester functional asymmetric α,ω-diene M1 in a 1:40 molar ratio in the presence of 4 mol% (to total acrylate groups) Hoveyda-Grubbs second generation catalyst. ¹H NMR (CDCl₃), δ (ppm): ¹H NMR (CDCl₃), δ (ppm): 7.53–7.37 (m, m-ArH), 7.20–6.89 (m, o-ArH + p-ArH + OCOCH=CH), 6.38– 6.33 (d, OCOCH= CH_2), 6.12-6.04 (m, OCOCH= CH_2), 5.78-5.67 (d, OCOCH=CH + CH₂=CHCH₂O), 4.80-4.76 (m, CH₂CHCH₂), 4.49-4.36 (m, $CH_2CH_2OCOCH=CH_2+CH_2=CHCOOCH_2CH_2+$ CH_2CHCH_2), 3.98-3.69 (m, PEG backbone + CH_2OPOCH_2), 3.34-3.16 (m, OCH₃), 2.91-2.82 (m, CH₂SCH₂CH₂), 2.68-2.39 (m, $CH_2SCH_2CH_2$), 2.01-1.78 (m, $CH_2=CHCH_2$), 1.53-1.37 (m, OCH₂CH₂CH₂), 1.35-1.03 (m, CH₂=CHCH₂(CH₂)₆). 31 P NMR, δ (ppm): 1.60. MALLS-GPC: $M_n = 34\ 200$, $M_w/M_n = 1.41$; ¹H NMR: $M_{\rm n} = 33~900.$

Results and discussion

In this article, the linear-dendron-like **l-Dm** block copolymers were investigated based on the second synthetic strategy utilizing the end group of a linear polymer chain. As already mentioned, the approach to the synthesis of **l-Dm** block PPEs via ADMET polymerization involves the reaction of phosphoester functional unsymmetric α,ω-diene containing both a terminal double bond and an acrylate (M1). The reaction will thus proceed by head-to-tail ADMET polymerization. A similar approach was used by Meier et al. for the synthesis of starshaped (block) copolymers, 16,20c in which polymerization was terminated with multifunctional (tri-, and tetra-) acrylates as selective chain stoppers since acrylate would react only with one chain end of the polymer (acrylates have a very low tendency to dimerize during olefin metathesis).18 For this purpose, we first prepared a linear precursor, the acrylated poly(ethylene glycol) methyl ether (**PEG-Acrylate**, $M_{n,NMR} = 1900$, $M_{n,MALLS} = 2000$,

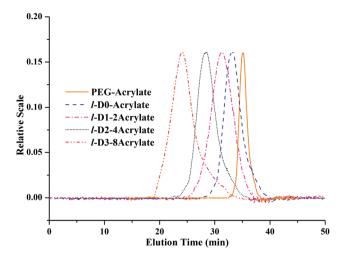


Fig. 1 Representative multiangle laser light scattering-gel permeation chromatography chromatograms for the linear macromolecular chain stopper and I-D*m* block PPEs.

 $M_{\rm w}/M_{\rm n}=1.05$, Fig. 1 and S2†), which may be considered a selective macromolecular chain stopper by ADMET polymerization of **M1**. Varying the **M1**: **PEG-Acrylate** molar ratio allowed the synthesis of different molecular weights of the B block in a similar fashion as with the variation of the [monomer]/[initiator] ratio in living/controlled polymerization reactions. The reaction of **M1** and **PEG-Acrylate** in a 5:1 molar ratio yielded a linear diblock PPE containing an acrylate end group, denoted as **l-D0-Acrylate**.

The analysis of **l-D0–Acrylate** by ¹H NMR (Fig. 2a) showed new formed internal α,β -unsaturated ester functions at 6.99 (x)

and 5.88 ppm (y), and the internal olefin peak of the unsaturated polyolefin at 5.38 ppm was not observed, indicating that ADMET polymerization between terminal olefins did not happen. We can also observe the appearance of resonance peaks at 6.42 ppm (a) and 6.23 ppm (c), which can be ascribed to the protons of the end acrylate group. Additionally, in the ¹³C NMR spectrum of **l-D0-Acrylate** (Fig. S3a†), as expected, two new peaks appeared at about 143.7 (Y) and 122.6 (X). These were in full agreement with the diblock copolymer structure. The average degree of polymerization (DP) was calculated from the end-group analysis of the ¹H NMR spectrum and was derived from the following formula: $DP = n = (S_f/2)/(S_g/3)$, which was used to determine the number-average molecular weight of I-**D0-Acrylate**, $M_{n,NMR} = [(S_f/2)/(S_q/3)] \times M_{(M1)} (424) + M_{(PEG-Acry-1)} + M_{(M1)} (424) + M_{(M1$ L_{late} (1900) - $M_{(ethylene)}$ (28) = 3900. The typical MALLS-GPC curve of the as-obtained I-D0-Acrylate showed a monomodal peak with a moderate molecular weight $(M_{n,MALLS})$ and a relatively narrow polydispersity (M_w/M_p) , as shown in Fig. 1 and Table 1. The molecular weight obtained was correlated with the value calculated by the ¹H NMR spectrum.

These results clearly demonstrate that it is possible to prepare a diblock copolymer *via* ADMET by growing a second block "step-by-step" from this macromolecular selective chain stopper. Certainly, the ongoing ADMET polymerization is a mixture of step-growth chemistry of **M1** (homo-oligomerization) and simultaneous chain-growth of **M1** to the formed block copolymer. However, the ¹H NMR analysis of the polymer revealed a head-to-tail metathesis selectivity of >98%. This high selectivity and the efficiency of the underlying cross metathesis reaction allow a complete coupling of possibly formed dimers and trimers (which were not observed by MALLS-GPC) to the

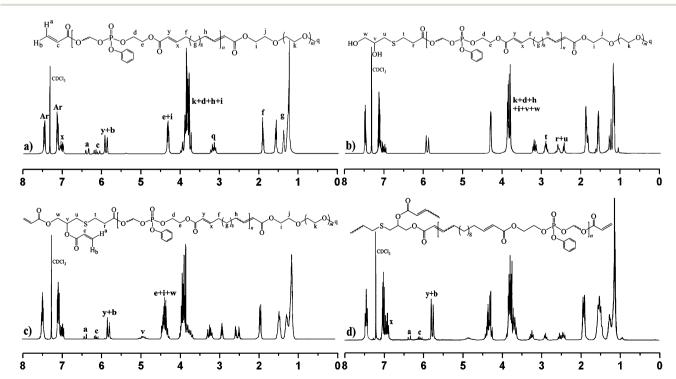


Fig. 2 ¹H NMR spectra for (a) l-D0-Acrylate, (b) l-D0-2OH, (c) l-D0-2Acrylate, and (d) l-D1-2Acrylate block PPEs.

Table 1 Characteristics of linear-dendron-like block polyphosphoesters via ADMET polymerization with selective chain stoppers^a

Polymer	$Yield^{b}$ (%)	[M]/[CS]	$M_{ m n,MALLS}^{c}$	$M_{\rm w}/M_{ m n}^{\ \ c}$	$M_{ m n,NMR}{}^d$	$\mathrm{DP_{n,NMR}}^e$
l-D0–Acrylate	93	5:1	4100	1.31	3900	4.8
l-D1-2Acrylate	92	10:1	8500	1.36	8300	10.5
l-D2-4Acrylate	93	20:1	17 100	1.37	16 800	20.3
l-D3-8Acrylate	91	40:1	34 200	1.41	33 900	40.9

 $[^]a$ Reaction conditions for preparation of linear-dendron block PPEs: polymerization temperature = 40 °C, polymerization time = 24 h, and [M]/[C] = 5:1-40:1. b Obtained gravimetrically from the dried polymer. c Both the actual molecular weight $(M_{n,MALLS})$ and the polydispersity (M_w/M_n) of I-Dm block PPEs were determined by the MALLS-GPC analysis. d $M_{n,NMR} = [(S_f/2)/(S_q/3)] \times M_{(M1)} + M_{(PEG-Acrylate)} - m \times M_{(ethylene)}$ was calculated by 1 H NMR spectroscopy, where $M_{(M1)} = 424$, $M_{(PEG-Acrylate)} = 1900$, and $M_{(ethylene)} = 28$ are the molar masses of the monomer, selective chain stopper and ethylene, respectively, m = 1, 3, 7, and $15.^c$ DP $_n = (S_f/2)/(S_q/3) - DP_{n-1} - DP_{n-2} - DP_{n-3}$ was obtained by 1 H NMR spectroscopy, where n = 1, 2, 3, and 4.

growing polymer. Therefore, overall, the polymerization behaves as a chain-growth polymerization would.¹⁶

The generated I-D0-Acrylate contained the acrylate end group, allowing for the direct functionalization of this linear macromolecule.27,28 The thiol-ene reaction is an advantageous tool because it is a rapid and quantitative reaction that achieves high conversion with little to no photoinitiator under atmospheric conditions, and it is insensitive to water and oxygen. 28-30 Acrylate-functionalized I-D0 is known to react rapidly with thiols by the thiol-Michael addition click reaction.31 Therefore, we reacted 1-D0-Acrylate with 3-mercapto-1,2-propanediol in the presence of triethylamine catalyst. The reaction proceeded at room temperature until complete and **1-D0** block PPE with two hydroxyl terminal groups (I-D0-2OH) was achieved. The esterification of I-D0-2OH with an excess of acryloyl chloride was then carried out at room temperature to produce 1-D0-2Acrylate. The structures, purity and molecular weights of 1-D0-2OH and I-D0-2Acrylate were affirmed by ¹H NMR (Fig. 2b and c) and ¹³C NMR spectra (Fig. S3b and c†), FT-IR spectra (Fig. S4†), and MALLS-GPC curves (Fig. S5†).

Subsequently, **l-Dm** (m = 1, 2, 3) block PPEs having 2, 4, and 8 branched chains were designed and synthesized via ADMET polymerization of M1 using different acrylate multifunctional macromolecular chain stoppers (as-obtained low-generation I-Dm), which were generated by successive thiol-Michael addition click reaction and esterification, as shown in Scheme 1. These I-**Dm** (m = 1, 2, 3) block PPEs were fully characterized by ¹H NMR, FT-IR, and MALLS-GPC, and the detailed results are summarized in Table 1. The typical MALLS-GPC curves of the resulting **l-Dm** block PPEs (Fig. 1) revealed the symmetrical elution peaks with relatively narrow polydispersity. Moreover, the actual molecular weights of these **l-Dm** block PPEs can be determined by ^{1}H NMR spectroscopy, and the $M_{n,NMR}$ s are consistent with $M_{\rm n,MALLS}$ s indicated that the molecular weights of these **l-Dm** block PPEs can be accurately controlled by the molar ratio of M1 to macromolecular chain stoppers.

As a representative example (Fig. 2d), the ¹H NMR spectrum of as-synthesized **I-D1** block PPE clearly shows that, besides the original proton signals of the PEG chain and acrylate end groups, new signals at 6.99 ppm (doublet of triplets) typical of internal acrylates clearly appeared. Meanwhile, the ¹H NMR and ¹³C NMR spectra of other **I-D***m* block PPEs gave similar results (*e.g.*, **I-D3**; Fig. S6†). These results demonstrated that **I-D***m* block

PPEs with terminal acrylate groups really played the role in the controlled ADMET polymerization of **M1**, and the dendron-like **l-Dm** with different branch densities can be easily synthesized according to Scheme 1.

Amphiphilic block copolymers are able to self-assemble into a variety of stable nanostructures when dissolved in a selective solvent due to the association of the insoluble core-forming blocks. Aggregate morphologies of amphiphilic block copolymers are closely correlated with the copolymer composition and the chain length ratio of the hydrophilic blocks to hydrophobic blocks. In this study, these amphiphilic block copolymers with linear-dendron-like architecture self-assembled into nanoparticles in a selective solvent at room temperature. The linear PEG block is totally soluble in water, while the dendritic PPE block is insoluble, and precipitated in water. The solubility discrepancy between the PPE and PEG blocks in the copolymer provided a possibility of self-aggregation to form nanoparticles in the selective solvent of water. Thus the aggregates were prepared by adding water into the copolymer solution of THF.

Both the morphology and the average size of the selfassembled nanoparticles from these diblock copolymers were

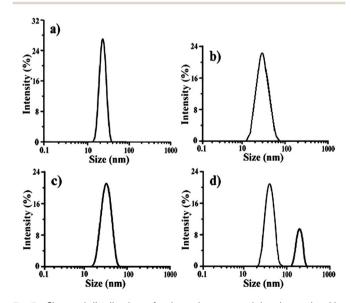


Fig. 3 Size and distribution of polymeric nanoparticles determined by means of DLS (a) l-D0, (b) l-D1, (c) l-D2, and (d) l-D3 block PPEs.

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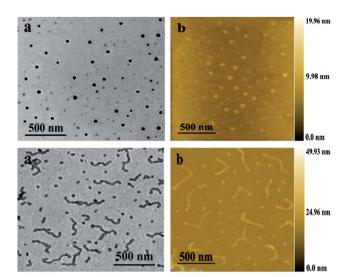


Fig. 4 The aggregate morphologies formed by the self-assembly of l-Dm (m=0 and 3) block PPEs with a concentration of 0.005 mg mL $^{-1}$ in THF–water: (a) TEM photographs and (b) AFM images.

Table 2 Self-assembled nanoparticles from the linear-dendron-like block polyphosphoesters in a selective solvent

Polymer	Average diameter ^a (nm)	PDI^b
l-D0–Acrylate	24.6 ± 3.7	0.166
l-D1–2Acrylate	35.9 ± 2.6	0.181
l-D2–4Acrylate	40.5 ± 4.1	0.175
l-D3–8Acrylate	47.0, 276.0	0.228

^a The average diameter of nanoparticles was determined by the dynamic light scattering (DLS) technique. ^b PDI denotes the polydispersities of nanoparticles in a selective solvent (THF–water).

investigated by the techniques of DLS (Fig. 3), AFM (Fig. 4b), and TEM (Fig. 4a and S7†), and the detailed results are shown in Table 2. Except for the I-D3 sample, most diblock copolymers with different branch densities self-assembled into spherical micelles with an average diameter of less than 50 nm. The average diameters increased slightly with increasing branch densities. Notably, these micellar nanoparticles had a diameter similar to the conventional polymeric micelles usually with a diameter of 10-50 nm,33 suggesting that they had a dendron-like PPE core surrounded by a hydrophilic PEG corona. As for the highest generation I-Dm block PPE, a mixed morphology of spherical micelles and a smaller amount of worm-like micelles were presented in AFM and TEM images. DLS analysis displayed a bimodal distribution (Fig. 3d), corresponding to the single spherical micelles with an average diameter of 47 nm and the worm-like micelles with the largest even exceeding 500 nm. In all, these results indicate that the linear-dendron-like diblock copolymers mainly self-assembled into micelles with controllable morphology and size, and both the copolymer composition and the dendritic topology of the hydrophobic core had no apparent influence on the morphology of micelles. This also provides a convenient method to generate the nanoparticles having a lower hydrophilic component in aqueous solution.

Conclusions

A new linear-dendron-like **l-Dm** block PPE with construction of unsymmetrical topology was successfully synthesized via ADMET polymerization. We have shown that ADMET polymerization of a phosphoester functional asymmetric α, ω -diene monomer in the presence of suitable mono- and multifunctional chain transfer agents affords **l-Dm** (m = 0, 1, 2, 3) block PPEs. Each obtained low-generation I-Dm block PPE was readily converted, by thiol-Michael addition click reaction and esterification, to a new macromolecular chain stopper by subsequent ADMET polymerization for the preparation of high-generation l-**Dm** block PPEs. The molecular weight of these **l-Dm** block PPEs can be well controlled using this polycondensation approach through the monomer/chain stopper feed ratio. In all cases, the only purification that was used involved precipitation, thus affording very high purity materials. Moreover, these block PPEs could mainly self-assemble into spherical micelles with different sizes (an average diameter of less than 50 nm) in a selective solvent (THF-water), and the dendron generation of the PPE core had no apparent influence on the morphology and the average size of nanoparticles. Consequently, this provides a versatile strategy not only for the synthesis of biodegradable and amphiphilic block PPEs with linear-dendron-like architecture but also for the fabrication of biocompatible nanoparticles with a suitable size for biomedical applications.

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