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## Incomplete Atomic Arrangement in Crystals

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 $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>.  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and the oxide obtained by the electrolytical oxydation of Al ( $\gamma'$ -Al<sub>2</sub>O<sub>3</sub>) are averaged structures with regard to the cations. Both the  $\gamma$  and  $\gamma'$  oxides have the same oxygen-lattice but there is a distinct difference in the degree of incomplete arrangement of the cations.  $\gamma'$ -Al<sub>2</sub>O<sub>3</sub> and  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (Fe<sub>2</sub>O<sub>3</sub>) are intermediate cases between the amorphous state and the completely arranged state.

RYSTAL structures in which part of the ✓ atoms (ions) are distributed statistically (averaged structures) have been described for a few compounds. Thus, in Li<sub>2</sub>O. Fe<sub>2</sub>O<sub>3</sub>, the Li and Fe ions are distributed at random over the cation positions of the simple NaCl-arrangement. In a number of compounds with spinel structure, e.g. Mg<sub>2</sub>GaO<sub>4</sub>, the Ga-ions and half of the Mg-ions take equivalent positions, with an averaged distribution ("variate atom equipoints").2 An interesting type of averaged structure is shown by the high temperature modification of Ag<sub>2</sub>HgI<sub>4</sub><sup>3</sup> and related compounds, since here we deal with an averaged distribution of vacant positions and cations over a number of definite lattice positions.

Striking cases of a statistical atomic arrangement, partially differing from the types mentioned above, were found to exist with a number of oxides A<sub>2</sub>O<sub>3</sub> and will be discussed in the present note. Crystallographical details, to be published in Zeits. Krist., will be omitted.

We examined the closely related compounds γ-Al<sub>2</sub>O<sub>3</sub> and γ-Fe<sub>2</sub>O<sub>3</sub>, the instable cubic modifications of these oxides. From density relationships and considerations concerning ionic radii we believed both the structure proposed by Welo and Baudisch<sup>4</sup> and the one given by Thewlis<sup>5</sup> to be incorrect. γ-Fe<sub>2</sub>O<sub>3</sub> and γ-Al<sub>2</sub>O<sub>3</sub> were found to have a face-centered oxygen lattice; the unit cell contains 32 oxygen ions. The cations are distributed in a way statistically over

the 8+16 spinel positions in such a way that per unit cell on the average  $21\frac{1}{3}$  cations and  $2\frac{2}{3}$ vacant positions are distributed over the 24 definite positions.

Besides, we studied the structure of the Al<sub>2</sub>O<sub>3</sub> obtained by the electrolytical oxidation of aluminium. This Al<sub>2</sub>O<sub>3</sub> is closely related with γ-Al<sub>2</sub>O<sub>3</sub> but in its x-radiograms some of the lines of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> fail. We have called this oxide  $\gamma'$ -Al<sub>2</sub>O<sub>3</sub>. This  $\gamma'$ -Al<sub>2</sub>O<sub>3</sub> has a face-centered arrangement of the oxygen ions too. The unit cell contains 4 oxygen ions and on the average 23Al3+ ions. These cations are distributed in a way statistically over 12 positions, viz. 8 in which the Al3+ ion is surrounded by 4O2-ions and 4 in which the coordination number is 6; the former positions contain about 30 percent, the latter about 70 percent of the cations.

Thus  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and  $\gamma'$ -Al<sub>2</sub>O<sub>3</sub> belong to the large class of compounds with a cubic close packed anion lattice, containing the smaller cations in the interstices between the anion spheres. Such a close packing contains, per unit cell of 4 anions, 8 tetrahedron holes (surrounded by 4 anions) and 4 octahedron holes. In Li<sub>2</sub>O the tetrahedron holes are all occupied (CaF2type), in MgO, FeO, etc., and the Li<sub>2</sub>O. Fe<sub>2</sub>O<sub>3</sub> mentioned above, the octahedron holes are filled (NaCl-type), the unit cell containing in both cases 4 oxygen ions.

It is convenient to describe the structure of AgI, Ag<sub>2</sub>HgI<sub>4</sub> ( $\alpha$  and  $\beta$ ) in a similar way. In these compounds too the unit cell contains 4I-ions according to a face centered lattice. In AgI half of the tetrahedron interstices are occupied in such a way that the cubic symmetry is maintained. In Ag<sub>2</sub>HgI<sub>4</sub> one of these positions

<sup>&</sup>lt;sup>1</sup> E. Posnjak and T. F. W. Barth, Phys. Rev. 38, 2234

<sup>(1931).

2</sup> T. F. W. Barth and E. Posnjak, Zeits. f. Krist. A82, 325 (1932); cf. also F. Machatschki, ibid. 82, 348 (1932).

J. A. A. Ketelaar, Zeits. f. Krist. A87, 436 (1934); Zeits. f. physik. Chemie B26, 327 (1934).

L. A. Welo and O. Baudisch, Phil. Mag. (6) 50, 399 (1925).

<sup>&</sup>lt;sup>5</sup> J. Thewlis, Phil. Mag., (7) 12, 1089 (1931).

<sup>&</sup>lt;sup>6</sup> W. G. Burgers, A. Claassen and J. Zernike, Zeits. f. Physik 74, 593 (1932).

is vacant; however, the three remaining positions can be occupied in different ways. In  $\beta$ -Ag<sub>2</sub>HgI<sub>4</sub> (low temperature modification) the cations take three definite positions, which leads to a tetragonal (pseudocubic) symmetry; in  $\alpha$ -Ag<sub>2</sub>HgI<sub>4</sub> the three cations are distributed statistically over the 4AgI positions, and the symmetry is again a cubic one.

Now in  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>,  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> and  $\gamma'$ -Al<sub>2</sub>O<sub>3</sub> the cations are distributed over both tetrahedron and octahedron holes.

 $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> and  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> can be derived from the spinel structure; this structure involves a larger cell, containing 32 oxygen ions per unit. In normal spinels the cations are distributed according to a regular arrangement over 8 tetrahedron holes and 16 octahedron holes. (The unit cell contains 64 tetrahedron positions and 32 octahedron positions.) In the  $\gamma$ -oxides a number of these 24 positions are vacant (preferably octahedron positions), on the average  $2\frac{2}{3}$  per unit cell. The averaged structures of the  $\gamma$ -oxides and of α-Ag<sub>2</sub>HgI<sub>4</sub> are therefore of a similar type. There is a partial arrangement of the cations since only part of the available interstices are used. A number of cations and vacant places are distributed at random over these definite positions.

However, a distinct difference between the  $\gamma$ oxides and α-Ag<sub>2</sub>HgI<sub>4</sub> exists. The latter compound has a transition point: at 50° the averaged structure is converted to the regular arrangement of β-Ag<sub>2</sub>HgI<sub>4</sub> (low temperature modification). This transition indicates that the cations in α-Ag<sub>2</sub>Hg I<sub>4</sub> must have a large mobility, and move more or less freely in the lattice. The  $\alpha$ modification has actually a high electrolytical conductivity3 and is a cation conductor. Accordingly there is an intimate connection between this conductivity and the averaged structure: the cations (especially the Ag+ ions) jump frequently from one AgI-position to another, leaving each of them on the average \(\frac{1}{4}\) of the time vacant. One could say that the Ag+ ions, though partially arranged according to AgI-positions, are in a kind of fluid state in these lattices. If the temperature is lowered to the transition point, the Ag<sup>+</sup>-ions "crystallize" and take definite positions.

 $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> has no appreciable electrolytical conductivity in its existence region. The probability of a jump from one spinel position to another is obviously negligible. The case is therefore different from that of  $\alpha$ -Ag<sub>2</sub>HgJ<sub>4</sub>. The incomplete arrangement of the cations is not caused by their thermal agitation but by the impossibility or improbability of a simple arrangement in the cubic close packed O<sup>-</sup>-lattice. It is comparable with a partially *amorphous state* of these cations.

In  $\gamma'$ -Al<sub>2</sub>O<sub>3</sub> the Al-ions are distributed quite statistically over all interstices of the lattice; the only restriction is that averaged 70 percent of the Al-ions have chosen an Al<sup>6</sup> position (octahedron hole), 30 percent an Al<sup>4</sup> position. The arrangement of the cations is still more incomplete than in the  $\gamma$ -oxides, and the partial (spinel) arrangement of the latter fails. Only small deviations from the statistical distribution are present, i.e., the cations are practically entirely in an amorphous state. In this respect,  $\gamma'$ -Al<sub>2</sub>O<sub>3</sub> is actually one of the most striking averaged structures so far found.

y' and γ-Al<sub>2</sub>O<sub>3</sub> can be considered as intermediate cases between the amorphous and the totally arranged state. In \(\gamma'\)-Al2O3 only the anions are arranged regularly. The second step from the amorphous to the completely arranged state can be realized by a careful heating of  $\gamma'$ -Al<sub>2</sub>O<sub>3</sub> (900°C). Then  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is formed, and a partial arrangement of the cations occurs in the way described which leads to a spinel arrangement with a larger unit cell. A complete arrangement of the cations is finally realized when γ-Al<sub>2</sub>O<sub>3</sub> is heated above 1000° Then α-Al<sub>2</sub>O<sub>3</sub> is formed, i.e., the cubic close packed oxygen lattice is transformed into a hexagonal close packing. A simple complete arrangement of cations and anions in a ratio 2:3 is, for geometrical reasons, obtained easily in the latter class of symmetry: α-Al<sub>2</sub>O<sub>3</sub> has a rhombohedral cell with 4Al3+ and 6O--, in which the cations occupy 2/3 of the octahedron holes.

<sup>&</sup>lt;sup>7</sup> J. A. A. Ketelaar, Zeits. f. Krist. 80, 190 (1931).

<sup>&</sup>lt;sup>8</sup> H. Sachse, Zeits. f. Physik **70**, 539 (1931).
<sup>9</sup> An example of a "wholly fluid" state of the cations is probably α-AgI. However, the structure given recently by L. W. Strock, Zeits. f. physik. Chemie **B25**, 441 (1934), seems to be not quite clear.