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## 1 Stable Cobalt Nanoparticles and Their Monolayer Array as an <sub>2</sub> Efficient Electrocatalyst for Oxygen Evolution Reaction

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Supporting Information

ABSTRACT: Monodisperse cobalt (Co) nanoparticles (NPs) were synthesized and stabilized against oxidation via reductive annealing at 600 °C. The stable Co NPs are active for catalyzing the oxygen evolution reaction (OER) in 0.1 M KOH, producing a current density of 10 mA/cm<sup>2</sup> at an overpotential of 0.39 V (1.62 V vs RHE, no iRcorrection). Their catalysis is superior to the commercial Ir catalyst in both activity and stability. These Co NPs are also assembled into a monolayer array on the working electrode, allowing the detailed study of their intrinsic OER activity. The Co NPs in the monolayer array show 15 times higher turnover frequency (2.13 s<sup>-1</sup>) and mass activity (1949 A/g) than the NPs deposited on conventional carbon black (0.14 s<sup>-1</sup> and 126 A/g, respectively) at an overpotential of 0.4 V. These stable Co NPs are a promising new class of noble-metal-free catalyst for water splitting.

xygen evolution reaction (OER) is commonly referred to 27 as electrochemical oxidation of " $O^{2-}$ " to  $O_2$ . It is an 29 important half-cell reaction and is coupled with hydrogen 30 evolution reaction (HER) in a water-splitting cell for efficient 31 proton reduction and hydrogen (H<sub>2</sub>) generation. <sup>1-3</sup> As a 32 thermodynamically "up-hill" reaction that involves multielectron transfer, it requires the input of energy to drive its 34 completion. In order to lower the reaction energy barrier, an 35 efficient catalyst is needed to promote the four-electron 36 oxidation process and make OER proceed at low over-37 potentials. For this purpose, nanostructured iridium (Ir) and 38 ruthenium (Ru) have been chosen as the state-of-the-art OER 39 catalysts. 4-10 Recently, earth-abundant transition-metal oxides/ 40 hydroxides, <sup>11–21</sup> especially Co-based complexes, <sup>22–25</sup> Co–O-41 based thin films, <sup>26–29</sup> nanoparticles (NPs), <sup>30–39</sup> and layered 42 structure, 40-45 are also explored as promising alternative 43 catalysts for OER. However, these oxides have generally low 44 electronic conductivity, limiting their potential in electro-45 catalysis enhancement.

One recent strategy applied to improve the OER catalytic 47 activity of these oxides is to couple them with a conductive 48 support such as graphene, <sup>17</sup> carbon nanotubes, <sup>14,33</sup> metallic 49 Au, 26,46 or even Au NPs, as demonstrated in the core/shell Au/

Co<sub>3</sub>O<sub>4</sub> structure.<sup>35</sup> These studies indicate that catalytic activity 50 may be further enhanced if the catalyst can be made more 51 electron conductive. Studying the general design of the 52 catalysts, we see that these conductive supports can only offer 53 a partial solution to the desired enhancement in catalysis, as the 54 electron transfer required for the OER on the catalytically 55 active surface may only be possible at the catalyst-support 56 interface, not on the entire catalyst surface. To further enhance 57 the OER activity, the catalyst itself is better conductive to 58 facilitate electron transfer for the OER.<sup>47</sup> This makes metallic 59 Co NPs a promising catalyst choice. However, it is known that 60 metallic Co NPs are chemically unstable, subject to fast 61 oxidation when exposed to air or in an oxygenated solvent. It is 62 therefore essential to stabilize metal Co NPs first before they 63 can be studied for the OER. Herein, we report that when 64 synthesized and treated properly, metallic Co NPs can be 65 stabilized to serve as a new class of efficient catalyst for OER in 66 0.1 M KOH. Moreover, using a water-air interface self- 67 assembly method, we have assembled these Co NPs into a well- 68 defined monolayer array. The uniform array of Co NPs allows 69 the detailed evaluation of their intrinsic activity for OER. The 70 monolayer catalyst shows about 15 times higher turnover 71 frequency (TOF) and mass activity than the Co NPs deposited 72 on conventional carbon support.

The Co NPs were synthesized through the thermal 74 decomposition of cobalt carbonyl  $[Co_2(CO)_8]$  in 1,2,3,4-75 tetrahydronaphthalene solution with oleic acid and dioctyl- 76 amine as surfactants (see Supporting Information (SI) for 77 experimental details). Figure S1 shows a typical trans- 78 mission electron microscopy (TEM) image of the as- 79 synthesized Co NPs that are monodisperse with the diameter 80 of 10  $\pm$  1 nm. The as-synthesized Co NPs were loaded on 81 Ketjen carbon (C) at an initial mass ratio of 1:1 through 82 sonication of Co NP dispersion and C support in hexane, 83 denoted as C-Co NPs (Figure 1A). Inductively coupled 84 f1 plasma atomic emission spectroscopy analysis confirmed the 85 mass ratio of Co NPs in the C-Co catalyst was 42%. As 86 demonstrated previously, 48 the as-synthesized magnetic Co 87 NPs are not stable and subject to fast oxidation. To improve 88 their stability, these C-Co NPs were annealed in Ar + 5%  $H_2$  at 89

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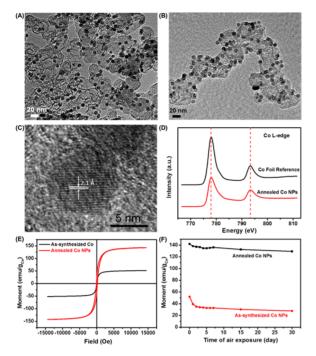


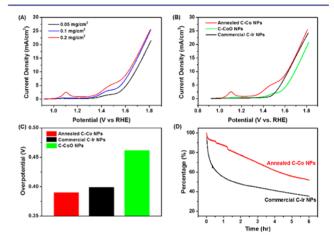
Figure 1. TEM image of (A) the as-synthesized C–Co NPs and (B) the C–Co NPs after reductive annealing in Ar + 5%  $\rm H_2$  at 600 °C for 1 h. (C) HR-TEM image of a representative single Co NP from (B). (D) Ex situ XAS spectra of Co L-edge of the Co reference foil and the annealed C–Co. (E) Room-temperature hysteresis loops of the C–Co NPs before and after reductive annealing. (F) Change of magnetic moment of the Co NPs vs the time of air exposure at room temperature.

90 600 °C for 1 h. This reductive annealing reduces the surface 91 oxide to metallic Co, increases the Co crystallinity, and removes 92 the surfactants to activate the catalyst for the subsequent 93 electrochemical measurements. The TEM image of the 94 annealed C—Co NPs in Figure 1B indicates no aggregation 95 of the Co NPs on the carbon support. High-resolution TEM 96 (HR-TEM) image of a representative Co NP deposited on C in 97 Figure 1C confirms that the annealed Co NP is indeed well-98 crystallized with the lattice fringe spacing measured to be 2.1 Å, 99 which is close to the (111) interplanar spacing of the face-100 centered cubic (fcc) Co. X-ray diffraction pattern (Figure S2) 101 further reveals that the crystallinity of the Co NPs is 102 significantly enhanced after reductive annealing.

Ex situ X-ray absorption spectroscopy (XAS) was applied to 104 study the chemical states of the annealed Co NPs. The XAS 105 measurements were performed in the total electron yield 106 detection mode to make them sensitive to the surface of the Co NPs. The XAS spectrum of the Co L-edge of the annealed Co NPs in Figure 1D shows two peaks at 778.0 and 793.3 eV that 109 match those for the metallic Co reference foil, indicating that 110 the surface of the annealed Co NPs is in metallic nature without 111 obvious surface oxidation. Chemical stability of the metallic C-112 Co NPs was monitored by the change of their magnetic properties using a vibrating sample magnetometer. Figure 1E is 114 room-temperature magnetic hysteresis loops of the C-Co NPs 115 before and after reductive annealing in Ar + 5% H<sub>2</sub> at 600 °C 116 for 1 h. The magnetic moments were normalized to the mass of 117 Co. Due to the surface oxidation, the initial Co NPs on the 118 carbon support are super paramagnetic with the magnetic 119 saturation moment  $(M_s)$  of 52 emu/g. After the reductive 120 annealing, their  $M_s$  increases to 142 emu/g, which is close to

the bulk Co value ( $\sim$ 162 emu/g). The  $M_{\rm s}$  change of the Co 121 NPs exposed to air was monitored, as shown in Figure 1F. The 122  $M_{\rm s}$  of the as-synthesized Co NPs drops to 39 emu/g after 1 day 123 and further decreases to 28 emu/g after 30 days (46% loss). As 124 a comparison, the annealed Co NPs are much more stable with 125 their  $M_{\rm s}$  staying at 129 emu/g level even after 30 days, 126 indicating that annealing indeed helps to stabilize Co NPs 127 against air oxidation. This stability enhancement is also more 128 significant than that for crystalline bcc-Fe NPs reported 129 previously.  $^{50,51}$ 

The OER activity of the annealed Co NPs was evaluated in  $^{131}$   $O_2$ -saturated 0.1 M KOH solution using a standard three- $^{132}$  electrode system. The C–Co NP catalyst was cast onto the  $^{133}$  glassy carbon (GC) working electrode. The OER polarization  $^{134}$  curves were recorded by linear sweep voltammetry at the scan  $^{135}$  rate of  $^{13}$  mV/s and continuous rotating speed of  $^{16}$ 00 rpm.  $^{136}$  Figure 2A shows the polarization curves of the annealed C–Co  $^{137}$  f2



**Figure 2.** (A) Polarization curves of (A) the annealed C–Co NPs on GC electrode at three different mass loadings and (B) the annealed C–Co NPs, C–CoO NPs and commercial C–Ir catalyst on GC electrode with a mass loading of 0.2 mg/cm². Measurements were performed on GC electrode in 0.1 M KOH at the scan rate of 10 mV/s and rotating speed of 1600 rpm. All the polarization curves were collected without *iR*-correction. (C) Comparison of the overpotential for different catalysts at the current density of 10 mA/cm². (D) Chronoamperometric curves of the annealed C–Co NPs and commercial C–Ir catalyst on GC electrode at an overpotential of 0.4 V (1.63 V vs RHE) in 0.1 M KOH.

catalyst with three different mass loadings without iR- 138 correction. The overpotential at the current density of 10 139 mA/cm², which is normally used for evaluating the electro- 140 chemical activity of an OER catalyst, 11,52 decreased from 0.45 141 to 0.39 V while increasing the catalyst loading from 0.05 mg/ 142 cm² to 0.2 mg/cm². Further increasing the mass loading 143 resulted in thicker catalyst film, causing limited mass transport 144 and detachment of catalyst from the electrode during 145 electrochemical measurement. Thus, the optimal catalyst 146 loading of 0.2 mg/cm² was chosen for the further study.

To better evaluate the OER activity of the annealed C-Co 148 NPs catalyst, the commercial Ir catalyst (10 wt % Ir on Vulcan 149 carbon black from Premetek Co., Figure S3) with the same 150 mass loading of 0.2 mg/cm² was chosen as a reference. The C- 151 CoO NPs were also studied as a control (see the SI and Figure 152 S4). Their polarization curves are shown in Figure 2B. For the 153 annealed Co NPs, during electrochemical oxidation, the surface 154 metallic Co atoms were oxidized, as indicated by the anodic 155

156 peaks observed at  $\sim$ 1.10 V for oxidation of  $Co^{II}$  to  $Co^{III}$  and 157 1.46 V for Co<sup>III</sup> to Co<sup>IV</sup>. However, the metallic core of the 158 annealed Co NPs increased their electronic conductivity and 159 thus enhanced the catalytic activity. Compared with the 160 insulating CoO NPs, the annealed Co NPs exhibited much 161 higher current density at all potentials and much smaller overpotential at the current density of 10 mA/cm<sup>2</sup> (0.39 vs 0.46 163 V). The OER activity of the annealed Co NPs is also better 164 than the commercial Ir catalyst. Their overpotentials at the 165 current density of 10 mA/cm<sup>2</sup> are summarized in Figure 2C. It 166 is worth mentioning that all the polarization curves shown here 167 are without any iR-correction. If considering the uncompensated resistance of the electrochemical cell, the overpotential of the annealed Co NPs for producing a current density of 10  $mA/cm^2$  decreased to ~0.3 V after iR-correction (Figure S5), 170 which is comparable or even smaller than the values of some 171 other nonprecious catalysts under similar experimental conditions. 14,16,35,42 172 173

The stability of the annealed C-Co NPs and commercial C-175 Ir NPs was tested by using a chronoamperometric method at an 176 overpotential of 0.4 V (Figure 2D). After 1 h, the current density for the annealed C-Co NPs dropped 13%, while the value from the C-Ir NPs decreased 45%. Even after 6 h test, the C-Co NPs still showed a slower rate of activity decrease than the commercial C-Ir, indicating that the annealed Co NPs are much more stable than the commercial Ir NPs. The NPs in the catalysts were further characterized after the stability test (Figure S6). The Ir NPs are aggregated, while the Co NPs stay well dispersed on the carbon support but have a core/shell 185 structure due to surface oxidation of Co, which explains their 186 slow activity drop over time. We should note that the oxidized 187 Co can be readily reduced back to Co by the same reductive 188 annealing described above and the C-Co catalyst can be regenerated and reused for the OER.

Previous studies have indicated that decreasing the thickness 191 of the catalysts can increase the number of active sites and 192 catalyst's TOF. 18,26 Very recently, self-assembled monolayer or 193 multilayer NP catalysts have been developed for enhanced 194 activity. 53-55 To better evaluate the OER catalysis of the Co 195 NPs, we assembled them into a monolayer array through a 196 water—air interface self-assembly approach. 56–58 The mono-197 layer array could be transferred easily to a carbon-coated TEM grid for TEM analysis (Figure 3A) or to a GC plate  $(0.6 \times 0.7)$ cm) for scanning electron microscopy (SEM) imaging (Figure 200 S7). The monolayer Co NP array on GC plate was preannealed at 600 °C as described and was used as a working 202 electrode; note: no NP aggregation was observed after the annealing treatment. From the TEM/SEM images of the Co NP array (Figures 3A and S7) we estimated the Co NP packing density at  $\sim 4000 \text{ NPs}/\mu\text{m}^2$ . The Co NP array showed high OER activity (Figure 3B) (the GC plate itself was not active for OER). The TOFs were calculated by assuming that all the surface Co atoms of the 10 nm fcc-Co NPs are catalytically active (see the SI). Figure 3C shows the TOFs at an overpotential of 0.4 V for the Co NP array and the C-Co catalyst with different mass loadings. For the catalyst with a  $_{212}$  mass loading of 84  $\mu g_{Co}/cm^2$  (C–Co NP catalyst at the loading 213 of 0.2 mg/cm<sup>2</sup> with 42 wt % Co), the TOF is 0.14 s<sup>-1</sup> 214 However, by decreasing the thickness of the catalyst to the 215 monolayer level (1.1  $\mu$ g<sub>Co</sub>/cm<sup>2</sup>), the TOF increases by about 216 15 times to 2.13 s<sup>-1</sup>. This TOF enhancement is in good 217 agreement with previous reports about the thickness effect on 218 the TOFs. 18,26 Moreover, the mass activity of the monolayer

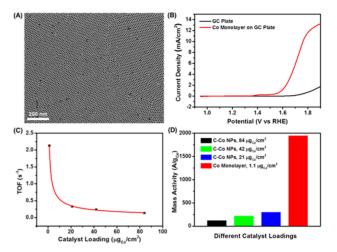


Figure 3. (A) TEM image of the monolayer assembly of Co NPs. (B) Polarization curves of the bare GC plate and the annealed Co NP monolayer catalyst on the GC plate at the scan rate of 10 mV/s in 0.1 M KOH. (C) TOFs and (D) mass activity of the annealed Co NP catalysts with different catalyst loadings at an overpotential of 0.4 V.

catalyst was also evaluated (Figure 3D). Compared to the C- 219 Co NP catalyst at a mass loading of 84  $\mu$ g<sub>Co</sub>/cm<sup>2</sup>, the 220 monolayer catalyst exhibits a dramatic enhancement in mass 221 activity from 126 to 1949 A/g. This value is also much higher 222 than that of the commercial C-Ir catalyst (500 A/g).

In summary, highly stable metallic Co NPs have been 224 synthesized through a post-reductive annealing process. These 225 metallic Co NPs are more active and durable for OER than the 226 commercial Ir catalyst. The high catalytic efficiency can be 227 ascribed to better electron conductivity of the metallic core. To 228 better evaluate the intrinsic activity of the metallic Co NPs, a 229 monolayer array of Co NPs has been fabricated through Co NP 230 self-assembly at water-air interface. The monolayer NP catalyst 231 exhibits 15 times higher TOF and mass activity than the Co 232 NPs deposited on conventional carbon support. With better 233 activity and stability than the commercial Ir catalyst, these 234 metallic Co NPs should serve as a promising noble-metal-free 235 catalyst for efficient OER in alkaline media.

#### ASSOCIATED CONTENT

#### Supporting Information

Experimental details, TOF, and mass activity calculations and 239 Figures S1-S7. The Supporting Information is available free of 240 charge on the ACS Publications website at DOI: 10.1021/ 241 jacs.5b04142. 242

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