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# The photocathodic properties of a Pb(Zr<sub>0.2</sub>Ti<sub>0.8</sub>)O<sub>3</sub> wrapped CaFe<sub>2</sub>O<sub>4</sub> layer on ITO coated quartz for water splitting†

Xiaorong Cheng, Deliang Chen, Wen Dong, Fengang Zheng, Liang Fang and Mingrong Shen\*

Pb(Zr<sub>0.2</sub>Ti<sub>0.8</sub>)O<sub>3</sub> wrapped CaFe<sub>2</sub>O<sub>4</sub> particles were constructed on ITO coated quartz as a photocathode for efficient water splitting. A photocurrent of 152 μA cm<sup>-2</sup> was obtained under zero bias vs. Ag/AgCl and 100 mW cm<sup>-2</sup> with the assistance of positive poling and Ag decoration.

Since Fujishima et al. reported n-type TiO<sub>2</sub> as photocatalyst for water splitting, photoelectrodes based on semiconductor materials have been extensively studied. Metal oxides with a narrow band gap such as WO<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub>, BiVO<sub>4</sub>, and TaON, have attracted much attention for their low cost, visible light activity and stability in electrolyte. However, they are mostly n-type and studied as photoanodes. Research on metal oxide photocathodes is quite sparse. Examples of the latter include Cu<sub>2</sub>O, Rh doped SrTiO<sub>3</sub>, and CaFe<sub>2</sub>O<sub>4</sub> (CFO).<sup>6-8</sup> Still, these materials do not provide stability combined with a good spectral photoresponse and high energy conversion efficiency needed for practical water splitting. p-Type CFO with a band gap of 1.9 eV may be a good potential photocathode material. Ida et al. reported a CFO photocathode prepared on a Pt substrate having a short circuit photocurrent of 200 µA cm<sup>-2</sup> under 500 W Xe lamp illumination when connected with an n-type TiO2 electrode.8 Unfortunately, this CFO electrode was prepared at a high annealing temperature of 1200 °C, making it impossible be prepared on most transparent conduction substrates like ITO/glass.

Ferroelectric films with spontaneous and reversible polarization, such as Pb(Zr,Ti)O<sub>3</sub>, <sup>9</sup> BiFeO<sub>3</sub>, <sup>10</sup> and BiFeCrO<sub>6</sub>, <sup>11</sup> have been intensively studied as potential photovoltaic materials in recent years, where the depolarization field  $(E_{pi})$  originating from the electric polarization that exists throughout the whole film and the barriers on the film/ electrode interface dominate the photovoltaic properties. 12,13 Our recent studies revealed that the Pb(Zr<sub>0.2</sub>Ti<sub>0.8</sub>)O<sub>3</sub> (PZT) film fabricated on ITO/quartz can be used as a photocathode in aqueous electrolyte. 14 However, PZT films can only absorb UV light. Combining the

Department of Physics & Jiangsu Key Laboratory of Thin Films & Collaborative Innovation Center of Suzhou Nano Science and Technology, Soochow University, Suzhou 215006. China. E-mail: mrshen@suda.edu.cn

† Electronic supplementary information (ESI) available: Experimental procedure and other informations for the PCP photoelectrodes (S1-S11). See DOI: 10.1039/ PZT films with a narrow band gap semiconductor is supposed to be an efficient way to encompass the UV to visible light spectrum and enhance the photovoltaic efficiency of PZT films. 15 In this study, we try to combine PZT and CFO to form a photocathode on ITO coated quartz. CFO is used to absorb visible light and PZT to absorb UV light and separate the photogenerated carriers by a remnant depolarization field. A thin PZT layer was firstly crystallized on ITO/quartz to ensure a close contact between ITO and PZT, and subsequently an amorphous PZT (a-PZT) layer was deposited on PZT/ITO. CFO particles were firstly prepared through an ordinary calcination method under high temperature of 1050 °C, and then deposited on a-PZT/PZT/ITO by electrophoretic deposition (EPD). The loose CFO particles were then wrapped intimately by spin-coating another PZT layer on the top. Finally, the whole a-PZT/CFO/a-PZT/PZT layer was sintered at 650 °C. We called this layer a PZT/CFO/PZT (PCP) layer. Ag particles deposited on the film surface by photochemical reduction were used to enhance the photoelectrochemical (PEC) properties. The experimental details are presented in the ESI.†

Fig. 1a shows the loosely stacked CFO particles deposited by the EPD method on PZT/ITO. This is reasonable since the particles attracted each other only by an electrical field in this stage, and annealing under 650 °C is not sufficient to re-crystallize the CFO layer which can not be well crystallized below 1000 °C. In addition,

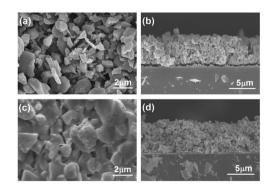


Fig. 1 The SEM surface (a) and cross-section (b) of the CFO layer deposited on PZT/ITO by the EPD method. (c) and (d) are the corresponding images for the PCP layer.

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the CFO layer is porous resulting from the irregular particle sizes of CFO powder. This CFO layer also loosely contacts the PZT/ITO/ quartz substrate, as demonstrated in Fig. 1b. An obvious gap can be seen between them. The poor connection among CFO and between CFO and the substrate leads to a poor photocurrent of about 1.5  $\mu$ A cm<sup>-2</sup> (Fig. S1, ESI†). We also found that the one with a PZT layer on the bottom or top also exhibits quite low photocurrent, because the former has a poor connection among CFO and the latter has a poor connection between the CFO and substrate (Fig. S2, ESI†).

The surface morphology of the PCP film is shown in Fig. 1c and d. The interparticle connection happens due to the PZT gel penetrating from the top of the CFO layer, which fills the voids between the CFO particles. A close contact between the CFO layer and substrate can also be observed in Fig. 1d. EDX analysis for the cross section of the PCP layer is performed and presented in the ESI† (Fig. S3). The PZT elements are dispersed throughout the PCP film, not only existing on the bottom and top of the layer. This means that PZT wraps the CFO particles in the whole layer (Fig. S4, ESI†). The CFO powder is well crystallized and represents an orthorhombic structure which is in accordance with ref. 16 and 17 and the PZT XRD pattern has a polycrystalline perovskite structure (JCPDS card No. 70-4260). The XRD pattern of Ag in the PCP film represents a pure Ag crystal phase (JCPDS card No. 65-2871) which confirms that Ag nanoparticles are formed on the PCP film surface by photoreduction (Fig. S5, ESI†).

As discussed above, the CFO layer without the bottom and top PZT coatings exhibits a very low photocurrent of about 1.5 μA cm<sup>-2</sup>. In contrast, the PCP film demonstrates a significant enhancement in the photocurrent to about 40 µA cm<sup>-2</sup>, indicating that the PZT wrapped structure is crucial to the separation and transport of light induced electron-hole pairs. When the PCP film was polarized by a positive applied voltage to the film *versus* the reference electrode,  $E_{\rm pi}$ is enhanced across the PZT with the direction pointing from layer/ electrolyte interface to ITO. In this case, the cathodic photocurrent of the PCP film increases to 80  $\mu$ A cm<sup>-2</sup>, since  $E_{\rm pi}$  is beneficial for the separation of photogenerated carriers. As expected, it decreases to about 5  $\mu$ A cm<sup>-2</sup> if the sample is poled by a reverse electric field. The photocurrent can be further enhanced to about 152 μA cm<sup>-2</sup> after the Ag is decorated on the film surface, as Ag decoration on the PZT surface can enhance the visible light utility and form a favorable energy level alignment for efficient electron-extraction and thus modify the film/electrolyte contact.14 Note there is an optimal deposition time for the photoreduction of Ag<sup>+</sup> (Fig. S6, ESI<sup>†</sup>).

In Fig. 2b, we also compare the current-voltage behavior of the as-prepared PCP film with the positively poled and positively poled plus Ag decorated ones. The zero-current potentials for the three samples are 0.21, 0.35 and 0.44 V vs. Ag/AgCl, respectively. Thus, both the positive poling and the Ag decoration make the open-circuit voltage of the photocathode enhance significantly, consistent with the enhancement of the photocurrent. Based on the current-voltage curve, the solar-to-chemical conversion efficiency for the positive poled and Ag decorated PCP film is determined to be nearly 0.1% (Fig. S7, ESI†).

To understand the increment of photocurrent, we measured in Fig. 3a the UV-vis diffused reflection spectra (DRS) of the PZT film, CFO and PCP layers, respectively. The absorption of CFO has a clear edge around 650 nm, which corresponds to the band gap of CFO.<sup>17</sup>

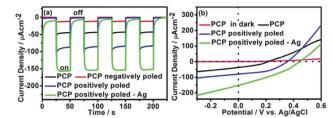


Fig. 2 (a) Photocurrent under zero voltage vs. Ag/AgCl measured in a PEC cell using a 0.1 M  $Na_2SO_4$  solution as the electrolyte and 100 mW cm<sup>-2</sup> illumination for the as-prepared (black) PCP film, positively (blue) or negatively (red) poled PCP film, and positively poled and Ag decorated (green) PCP film. (b) Current-voltage curves of samples shown in (a) except the negatively poled one. The one for the as-prepared PCP film in the dark is shown as pink, which is close to zero.

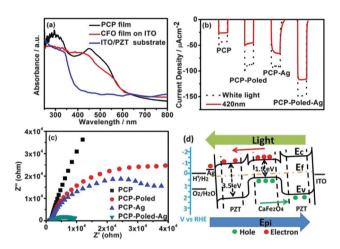


Fig. 3 (a) The UV-vis DRS spectra of PZT, CFO layer deposited by EPD and PCP film; (b) photocurrent density (solid lines) of different PCP films with wavelength cutoff filter of 420 nm and 0.1 M Na<sub>2</sub>SO<sub>4</sub> solution as electrolyte. The dotted lines are those measured without the filter; (c) EIS spectra of the different samples; (d) the energy band structure of the positively poled and Ag decorated PCP film.

The absorption edge around 400 nm of PZT film on ITO quartz comes from the band gap of PZT. The PCP film has two clear edges of CFO and PZT. This implies that the light-harvesting of PCP film is due to the narrow band gap of CFO and broad band of PZT that encompass the UV to visible light spectra, which converts into the photocurrent (Fig. S8, ESI†). Based on the absorption spectra, we measured the photocurrent using a wavelength cutoff filter of 420 nm in order to separate the contribution from the visible light where PZT is not excited. Fig. 3b shows that the photocurrent coming from the visible light (>420 nm) for the as-prepared PCP film is 22  $\mu$ A cm<sup>-2</sup>. When the film is positively poled, the visible light photocurrent increases significantly to 50 μA cm<sup>-2</sup>, implying that  $E_{pi}$  not only helps to separate the photocarriers generated in PZT, but also the ones generated in CFO. For the case of Ag decorated on the as-prepared PCP films, the visible light photocurrent also increases obviously to 50  $\mu A$  cm<sup>-2</sup>, meaning that Ag nanoparticles in contact on the PZT surface enhance the visible light utility, consistent with our previous study.14 When Ag is decorated on the polarized PCP film, the visible light photocurrent increses to 120  $\mu$ A cm<sup>-2</sup>. Clearly, the two factors,  $E_{\rm pi}$  and Ag decoration

are both important for the enhancement of photocurrent observed in Fig. 2a.

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The corresponding EIS measurement was also carried out under the same conditions as the photocurrent measurement. In Fig. 3c the samples show pronounced arcs at higher frequencies, whose diameters provide the reaction rate occurring at the surface of electrode. Based on the Nyquist plots, significant changes in the EIS are observed. Comparing with the as-prepared sample, the semicircle decreases significantly as the PCP film is positively polarized or decorated by Ag, indicating a more effective separation of photo-generated electron-hole pairs and fast charge transfer to the electron donor–electron acceptor on the film surface. Phe polarized plus Ag decorated PCP film shows the smallest semicircle, indicating it to be the best for the charge separation and transfer, consistent with the observed enhanced PEC properties shown in Fig. 2 and 3b.

According to the analysis in Fig. S5 and S9 of the ESI,† we proposed that the PZT wrapped CFO structure may contribute mainly to the visible-light photocurrent. A physical picture is therefore described in Fig. 3d to explain this point, where the energy band structure of a symmetrical PZT/CFO/PZT heterojunction is sketched. The built-in electric fields induced at the left and right PZT-CFO junctions have the same magnitude but opposite direction if there are no other internal electric fields in the film. They do not contribute to separation and transport of the photocarriers. However, the PZT film may be self-polarized, and also there is an interface barrier on ITO/PZT.13 This leads to the photocathodic current of the as-prepared film. If the film is positively poled, the electric field  $E_{pi}$  in the bulk region of a polarized PZT thin film modifies the potential of each energy level (i.e.  $E_c$ ,  $E_v$  and  $E_f$ ), and provides a more strong driving force to drive the photocarriers out from the CFO, <sup>20</sup> resulting in an obvious photocurrent in the polarized sample when high energy photons ( $\lambda$  < 420 nm) are filtered out.

The evolution of hydrogen gas during the water splitting reaction under illumination was measured and the photocurrent during the reaction was also recorded on an electrochemical workstation to calculate the amount of electrons passing through the outer circuit. Fig. 4 shows the time course of hydrogen evolution from the PCP film (cathode)–Pt (anode) electrode system in H<sub>2</sub> saturated acidic 1 M HClO<sub>4</sub> solution under zero voltage vs. Ag/AgCl. During the measurement, bubbles could be seen on the surface of the PCP film and Pt mesh. The amount of H<sub>2</sub> evolved is less than half of the

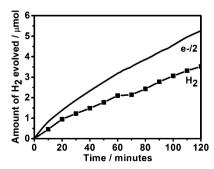


Fig. 4 Hydrogen evolution for the positively poled and Ag decorated PCP photocathode measured in 1 M HClO $_4$  solution as the electrolyte and 100 mW cm $^{-2}$  illumination.

electrons passing through the outer circuit. The Faradic efficiency for the  $H_2$  production reached nearly 80% in the first 20 minutes. Then, gas production rate decreased with increasing irradiation time. After 2 hours, the Faradic efficiency for the  $H_2$  production decreased to nearly 60%. The efficiency of  $H_2$  evolution was less than 100% and decreased with increasing irradiation, probably due to the unwanted backward reaction between  $H_2$  and  $O_2$ .<sup>17</sup> In addition, the stability issues of the photoelectrode may also cause the reduction of efficiency with time although the corrosion of electrode surface is not so evident (Fig. S9, ESI†). In spite of these limitations, the result clearly shows that the polarized PCP decorated with Ag nanoparticles can actually work as a photocathode. Note the  $O_2$  production is observed on the Pt anode (Fig. S10, ESI†).

In conclusion, we fabricated a PCP structure on ITO/quartz transparent substrates under 650  $^{\circ}$ C. The photocathodic properties can be significantly improved by the composite structure, poling the ferroelectric PZT and decorating the layer/electrolyte interface with Ag. A photocurrent of 152  $\mu$ A cm<sup>-2</sup> was obtained under zero bias  $\nu$ s. Ag/AgCl and 100 mW cm<sup>-2</sup> illumination, where 120  $\mu$ A cm<sup>-2</sup> is from the visible light (>420 nm).

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