

## Nanosheet thickness-modulated MoS<sub>2</sub> dielectric property evidenced by field-effect transistor performance†

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Sung-Wook Min,<sup>‡a</sup> Hee Sung Lee,<sup>‡a</sup> Hyoung Joon Choi,<sup>a</sup> Min Kyu Park,<sup>b</sup>  
Taewook Nam,<sup>c</sup> Hyungjun Kim,<sup>c</sup> Sunmin Ryu<sup>b</sup> and Seongil Im<sup>\*a</sup>

We report on the nanosheet-thickness effects on the performance of top-gate MoS<sub>2</sub> field-effect transistors (FETs), which is directly related to the MoS<sub>2</sub> dielectric constant. Our top-gate nanosheet FETs with 40 nm thin Al<sub>2</sub>O<sub>3</sub> displayed at least an order of magnitude higher mobility than those of bottom-gate nanosheet FETs with 285 nm thick SiO<sub>2</sub>, benefiting from the dielectric screening by high-*k* Al<sub>2</sub>O<sub>3</sub>. Among the top-gate devices, the single-layered FET demonstrated the highest mobility of  $\sim 170 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  with  $90 \text{ mV dec}^{-1}$  as the smallest subthreshold swing (SS) but the double- and triple-layered FETs showed only  $\sim 25$  and  $\sim 15 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  respectively with the large SS of 0.5 and  $1.1 \text{ V dec}^{-1}$ . Such property degradation with MoS<sub>2</sub> thickness is attributed to its dielectric constant increase, which could rather reduce the benefits from the top-gate high-*k* dielectric.

Graphene and related two-dimensional (2D) nanosheet materials have been studied with much attention in view of their applications to future nanoelectronics.<sup>1,2</sup> These 2D materials have initially been prepared by the exfoliation method.<sup>2,3</sup> Graphene has its high carrier mobility ( $\mu$ ) over  $100\,000 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  but also reveals intrinsic limitations caused by its rather small bandgap ( $E_g$ ).<sup>4–6</sup> Even though significant efforts to open up the bandgap of graphene were made modifying its form into a nano-ribbon, the gap turned out to be only around 200 meV while the intrinsic mobility was seriously reduced to  $\sim 200 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ .<sup>7,8</sup> Very recently, molybdenum disulfide (MoS<sub>2</sub>) layers appeared to overcome the dilemma of graphene as an alternative nanosheet material.<sup>9–15</sup> Although  $E_g$  of bulk MoS<sub>2</sub> is known to be  $\sim 1.2 \text{ eV}$  of indirect type, a few angstrom-thin single-layered MoS<sub>2</sub> has recently been reported to exhibit a direct bandgap of 1.8 eV, a

high mobility of  $200\text{--}350 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ , and a high on/off current ratio of  $10^6\text{--}10^8$  along with a very low subthreshold swing (SS) when used as the channel of a top-gate transistor with thin high-*k* oxide, which provides dielectric engineering for mobility enhancement.<sup>15–21</sup> However, as the nanosheet thickness increases, the mobility appears to decrease and the detailed mechanisms for mobility change due to the MoS<sub>2</sub> thickness still remain unclear.<sup>16</sup> In the present work, we demonstrated top-gate field-effect transistors (FETs) with single- to triple-layered MoS<sub>2</sub> nanosheets adopting a 40 nm thin atomic layer deposited (ALD) Al<sub>2</sub>O<sub>3</sub>, and also clarified the nanosheet thickness-involved issues: mobility and SS changes. Our nanosheet FET with a single-layered MoS<sub>2</sub> channel exhibited a maximum mobility of  $170 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ , which systematically decreases to 25 and  $15 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  with double- and triple-layered MoS<sub>2</sub> as respective channels.

Surface-cleaned 285 nm thick SiO<sub>2</sub>/p<sup>++</sup>-Si wafer was chosen as the substrate for our n-type MoS<sub>2</sub> nanosheet transistor with Au source/drain (S/D) electrodes, since 285 nm was reported as an optimal thickness to identify few-layered dichalcogenide.<sup>22,23</sup> Indeed, each exfoliated MoS<sub>2</sub> flake showed a distinctive optical contrast with thinner flakes exhibiting less optical density, which enabled fast screening of few layered-flakes under the optical microscope. Fig. 1(a) and (b) display a MoS<sub>2</sub> flake which was caught *via* photolithography to contact with patterned Au for S/D. Since the flake size was as long as  $\sim 10 \mu\text{m}$ , a  $5 \mu\text{m}$  long channel was available for our device (*W/L* ratio was  $\sim 1$ ). Fig. 1(c) illustrates a 3D scheme of our FET with a MoS<sub>2</sub> nanosheet and a 40 nm thin Al<sub>2</sub>O<sub>3</sub> dielectric, which was obtained by atomic layer deposition (ALD) at 200 °C. Fig. 1(d) shows Raman spectra obtained from many nanosheets assigned to single- to 6-layered MoS<sub>2</sub> flakes spanning  $\sim 5 \times 5 \mu\text{m}^2$ . As shown by the previous report,<sup>24</sup> the frequency difference between  $E_{2g}^1$  and  $A_{1g}$ , the two prominent Raman-active modes of 2H-MoS<sub>2</sub> crystals, increases stepwise with the number of layers. The inter-peak separation or frequency difference for the thinnest flake in Fig. 1(d) was  $\sim 18.0 \text{ cm}^{-1}$  which is in excellent agreement with that for single-layered MoS<sub>2</sub>.<sup>24</sup> Our results are also consistent with theoretical calculations based on the density functional perturbation theory.<sup>25</sup> Confirming the layer number of exfoliated flakes by Raman spectra,

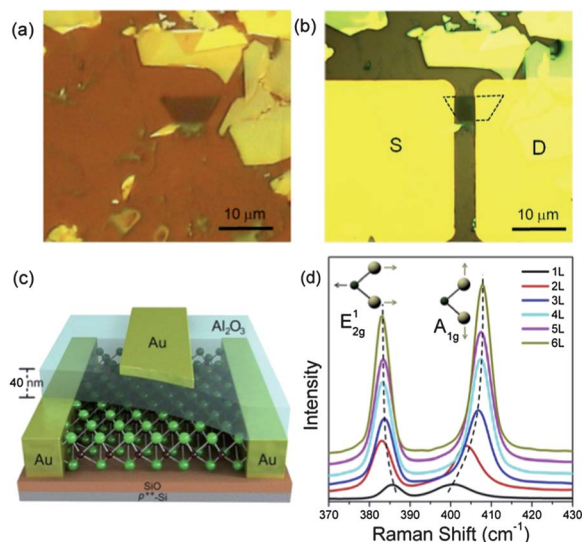
<sup>a</sup>Department of Physics, Yonsei University, Seoul 120-749, Korea. E-mail: semicon@yonsei.ac.kr; Fax: +82-2-392-1592; Tel: +82-2-2123-2842

<sup>b</sup>Department of Applied Chemistry, Kyung Hee University, Yongin, Gyeonggi 446-701, Korea

<sup>c</sup>School of Electrical and Electronic Engineering, Yonsei University, Seoul 120-749, Korea

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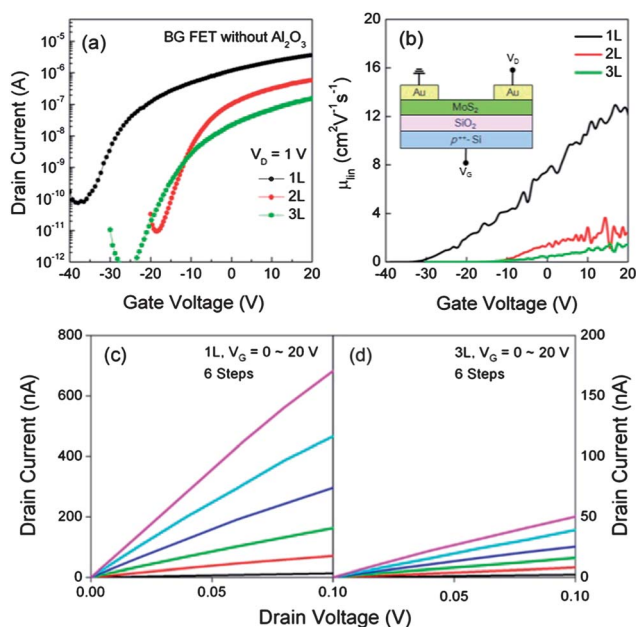
‡ These authors contributed equally to this work.



**Fig. 1** Optical microscopy images of (a) a few-layered MoS<sub>2</sub> flake on 285 nm thick SiO<sub>2</sub> and (b) the flake contacted with Au for S/D electrodes. (c) Three-dimensional schematic view of a transistor with a single-layered MoS<sub>2</sub> nanosheet. (d) Raman spectra of 1–6 layered MoS<sub>2</sub> flakes, along with inset figures for two representative Raman-active modes, E<sub>2g</sub><sup>1</sup> and A<sub>1g</sub>.

we selected single- to triple-layered MoS<sub>2</sub> as the channel for our nanosheet transistor. (The thickness of monolayer MoS<sub>2</sub> was measured by previous researchers to be  $\sim 0.65$  nm.)<sup>15</sup>

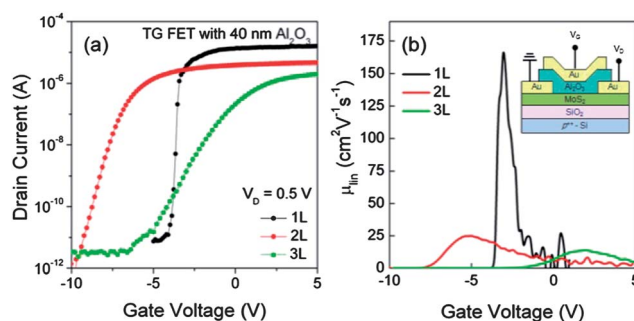
Fig. 2(a) and (b) respectively show the drain current–gate voltage ( $I_D$ – $V_G$ ) transfer curves and linear mobility ( $\mu_{lin}$ ) plots of bottom-gate FETs with a 285 nm thick SiO<sub>2</sub> dielectric (Fig. 2(b), inset) with the three different MoS<sub>2</sub> nanosheet channels: single, double, and triple layered MoS<sub>2</sub>. With the increase of the layer number, the drain



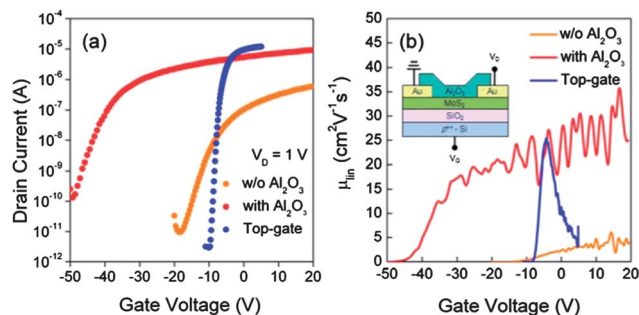
**Fig. 2** (a) The transfer and (b) linear mobility curves of bottom gate-controlled FETs with 1, 2, and 3 layered MoS<sub>2</sub>. The inset shows a 2D schematic of the devices. Output curves of bottom gate FETs of (c) single- and (d) triple-layered MoS<sub>2</sub> channel with Au S/D electrodes.

current ( $I_D$ ) and linear mobility ( $\mu_{lin}$ ) apparently decrease as shown in the figures. According to the mobility plots, the highest maximum mobility of the nanosheet FET was obtained from single layered MoS<sub>2</sub> to be  $\sim 13$  cm<sup>2</sup> V<sup>−1</sup> s<sup>−1</sup> but the mobility appears to be only  $\sim 1$  cm<sup>2</sup> V<sup>−1</sup> s<sup>−1</sup> with the triple layered MoS<sub>2</sub> channel, primarily because the inter-layer scattering of electrons reduces the mobility in the source-to-drain transport while such scattering would be subdued in the single-layered channel.<sup>26</sup> However, according to previous reports, these bottom-gate mobility results may not always be depending on the MoS<sub>2</sub> thickness but more sensitively depending on the density of local states between MoS<sub>2</sub> and SiO<sub>2</sub>, even though one tries to fabricate the bottom-gate MoS<sub>2</sub> FETs using the same experimental conditions; so a broad range of linear mobility has been reported for bottom-gate MoS<sub>2</sub> FETs.<sup>27–29</sup> The  $\mu_{lin}$  was estimated and plotted as a function of gate voltage ( $V_G$ ), based on the following equations:  $g_m(V_G) = dI_D/dV_G$  (transconductance) and  $\mu_{lin} = (g_m/C_{ox}V_D) \times (L/W)$ , where  $C_{ox}$  is the dielectric capacitance. Fig. 2(c) and (d) show that our single- to triple-layered MoS<sub>2</sub> channels have good ohmic contacts with Au S/D electrodes.

The linear mobility of the MoS<sub>2</sub> nanosheet FETs suddenly increases by at least an order of magnitude if equipped with our 40 nm thin Al<sub>2</sub>O<sub>3</sub> top-gate dielectric (see the inset of Fig. 3(b) for the top-gate FET scheme). According to the transfer curves of Fig. 3(a), the top-gate controlled nanosheet FETs display 12, 5, and 1.5  $\mu$ A of on-current  $I_D$  respectively with single, double, and triple layered MoS<sub>2</sub> under  $V_D = 0.5$  V. The mobility plots in Fig. 3(b) display 170, 25, and 15 cm<sup>2</sup> V<sup>−1</sup> s<sup>−1</sup> as the respective maximum linear mobilities of single, double, and triple layered MoS<sub>2</sub> FETs. Such highly enhanced mobilities and  $I_D$  current, which are at least an order of magnitude higher than those of bottom-gate devices, can only be explained by dielectric screening effects from the high- $k$  top-dielectric overlayer ( $k \sim 7$  in ALD Al<sub>2</sub>O<sub>3</sub>).<sup>21</sup> The on/off  $I_D$  ratio was  $5 \times 10^5$  to  $10^6$  for all cases. The high- $k$  overlayer effects were again confirmed by bottom-gate controlled FETs with double layered MoS<sub>2</sub> as respectively shown in the transfer curves and mobility plots of Fig. 4(a) and (b). According to the transfer curves in Fig. 4(a), the bottom-gate FET with an Al<sub>2</sub>O<sub>3</sub> overlayer displays apparently higher  $I_D$  by an order of magnitude than that of the other bottom-gate device without the top layer. The linear mobility of the bottom-gate controlled FET with the overlayer appeared almost similar to that of the top-gate control device to be 25–30 cm<sup>2</sup> V<sup>−1</sup> s<sup>−1</sup> (Fig. 4(b)), even though the SS values of top- and bottom-gate FETs are very different from each other.



**Fig. 3** (a) The transfer and (b) linear mobility curves of 1, 2, and 3 layered MoS<sub>2</sub> top gate FETs with a 40 nm thin Al<sub>2</sub>O<sub>3</sub> dielectric. The inset shows the 2D schematic of our top gate devices.



**Fig. 4** (a) The transfer and (b) linear mobility curves of double-layered MoS<sub>2</sub> bottom gate FETs with and without Al<sub>2</sub>O<sub>3</sub> overlayer as compared to the top gate FET with an Al<sub>2</sub>O<sub>3</sub> dielectric. The inset shows the 2D schematic of this bottom gate FET with an Al<sub>2</sub>O<sub>3</sub> overlayer.

As shown above, we could initially notice that the mobility of MoS<sub>2</sub> nanosheets is enhanced with the high-*k* overlayer due to the dielectric screening, whether the high-*k* overlayer plays as a gate dielectric or just an overlayer. Besides, the mobility of our FETs appears modulated and actually reduced with the MoS<sub>2</sub> nanosheet thickness whether the device is bottom-gate- or top-gate-controlled; it was primarily attributed to the thickness-induced (inter-layer) carrier scattering.<sup>26</sup> However, there exists another noticeable phenomenon which is also thickness-related: SS degradation with the nanosheet thickness. The thickness-induced SS degradation is particularly apparent with the top-gate MoS<sub>2</sub> FETs (Fig. 3(a)) and this should be properly explained along with the mobility reduction. According to a well-known theory, the SS value of FETs originates from an equation below:<sup>30,31</sup>

$$SS = (\ln 10)(kT/q) \left[ 1 + q x_{ox} \left( \sqrt{q^2 \epsilon_{ch} N_{bt} / kT} + q D_{it} \right) / \epsilon_{ox} \right], \quad (1)$$

where  $kT = 0.026$  eV at room temperature,  $\epsilon_{ch}$  and  $\epsilon_{ox}$  are the respective dielectric constants of the channel semiconductor and oxide dielectric,  $x_{ox}$  is the dielectric oxide thickness,  $N_{bt}$  and  $D_{it}$  are respectively the near-interface-bulk trap density and the interface trap density-of-states (DOS) at the channel/dielectric interface trap as energy-independent average values; their units are cm<sup>-3</sup> and cm<sup>-2</sup> eV<sup>-1</sup>, respectively. If we assume that the single-, double-, and triple-layered MoS<sub>2</sub>-Al<sub>2</sub>O<sub>3</sub> interfaces have almost the same and negligible interface trap DOS,  $D_{it}$ , above eqn (1) becomes mostly dependent on the following four factors:  $\sqrt{\epsilon_{ch}}$ ,  $\sqrt{N_{bt}}$ ,  $x_{ox}$ , and  $\epsilon_{ox}$ . (Since we prepared the different-layered MoS<sub>2</sub> flake samples in the same way, the process-contamination-induced  $D_{it}$  would be almost the same for all FETs.)

According to our Fig. 3(a), the SS of the top-gate FET with single-layered MoS<sub>2</sub> was as small as 90 mV dec<sup>-1</sup> while the SS decreases with the layer thickness or layer number (0.5 and 1.1 V dec<sup>-1</sup> for double and triple layers). It is very likely that  $\epsilon_{ch}$  is closely related to the MoS<sub>2</sub> thickness as the band gap does; larger the MoS<sub>2</sub> thickness, so its  $\epsilon_{ch}$  does become larger until the thickness reaches the bulk state.  $N_{bt}$  may also be dependent on the nanosheet thickness, possibly existing at the MoS<sub>2</sub> inter-layer locations. If  $\epsilon_{ch}$  and  $N_{bt}$  are irrelevant to the MoS<sub>2</sub> thickness and rather identical for all the MoS<sub>2</sub> nanosheets, the SS must be almost the same, which is, however, not matched to our experimental results at all. Interestingly, very recent

theoretical study supports the thickness-dependent  $\epsilon_{ch}$  which increases with MoS<sub>2</sub> layer thickness.<sup>25,32</sup> According to the theoretical results, dielectric constants of MoS<sub>2</sub> increase with the nanosheet thickness, which is opposite to the band gap behavior.<sup>16,20</sup> Unless the surface roughness of the single-, double-, and triple MoS<sub>2</sub> layers or the trap densities at the three MoS<sub>2</sub>-Al<sub>2</sub>O<sub>3</sub> interfaces are much different from one another, the thickness-dependent change of the nanosheet dielectric constant would be the main contribution able to explain the SS degradation.

Intrigued with the thickness-induced  $\epsilon_{ch}$  change, we have fabricated another set of top-gate nanosheet FETs with a high-*k* organic gate-dielectric, which used to be a 200 nm thick ferroelectric organic polymer [P(VDF-TrFE);  $k \sim 7$ ], to observe the MoS<sub>2</sub> thickness-dependent SS changes. According to the transfer curves of ferroelectric memory FETs with MoS<sub>2</sub> nanosheets (ESI, Fig. S1†), the SS of the ferroelectric memory FET with single-layered MoS<sub>2</sub> again appears to be smallest along with the largest memory window while it degrades with the nanosheet thickness that causes memory window decrease.<sup>33</sup> Unlike the cases of top-gate FETs, our bottom-gate nanosheet FETs in Fig. 2(a) and (b) did hardly show clear signs of such SS difference for single-, double-, and triple-layered MoS<sub>2</sub> FETs. It is probably because their SS values are already too large to show visible SS differences in view of eqn (1), according to which large SS values for all FETs are expected from low-*k* gate oxide ( $\epsilon_{SiO_2} = 3.9 \epsilon_0$ ) thickness ( $x_{ox} = 285$  nm). In spite of the bottom-gate structure, a very small SS value is achievable if we use high-*k* thin oxide as a gate dielectric, since that approach again meets the conditions for a small SS in eqn (1).<sup>27</sup>

As a final point, it could be worthwhile to understand that the mobility decrease of our top-gate MoS<sub>2</sub> nanosheet FETs in Fig. 3(b) is closely related to the dielectric constant increase with the MoS<sub>2</sub> thickness. Although the inter-layer scattering is already expected to be a generally known source of mobility loss in a low dimension channel, the dielectric constant increase with 2D nanosheet thickness could be another impelling one. If the increased  $\epsilon_{ch}$  competes with (high-*k*)  $\epsilon_{ox}$ , the source-to-drain electric field could not be confined to the channel thickness, and as a result, the high mobility-enabling mechanism by top-gate high-*k* dielectric engineering possibly becomes invalidated. Hence, the channel mobility is so closely related to the thickness of the MoS<sub>2</sub> nanosheet in the two aspects: inter-layer scattering and degradation of dielectric engineering (or degradation of SS).

## Conclusions

In summary, the nanosheet-thickness effects on the performance of MoS<sub>2</sub> FETs were studied. Our top-gate nanosheet FETs with 40 nm thin Al<sub>2</sub>O<sub>3</sub> displayed at least an order of magnitude higher mobility and smaller SS than those of bottom-gate nanosheet FETs with 285 nm thick SiO<sub>2</sub>, benefiting from the dielectric screening by high-*k* Al<sub>2</sub>O<sub>3</sub>. Among the top-gate devices, the single-layer FET demonstrated the highest mobility of  $\sim 170$  cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> with 90 mV dec<sup>-1</sup> as the smallest SS but the double- and triple-layered FETs showed only  $\sim 25$  and  $\sim 15$  cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> respectively with the large SS of 0.5 and 1.1 V dec<sup>-1</sup>. Such property degradations with MoS<sub>2</sub> thickness are actually attributed to the dielectric constant increase. The most superior device performance would be expected from a top-gate



single layered MoS<sub>2</sub> FET with a high-*k* dielectric, and the double- and triple-layered FETs could be promising only under a very high-*k* top-gate dielectric, of which the constant is even higher than that of thick MoS<sub>2</sub>.

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