## **ChemComm**



## COMMUNICATION

**View Article Online** 



Cite this: Chem. Commun., 2015 **51** 15125

Received 13th July 2015, Accepted 19th August 2015

DOI: 10.1039/c5cc05788d

www.rsc.org/chemcomm

## Ultrathin HNbWO<sub>6</sub> nanosheets: facile synthesis and enhanced hydrogen evolution performance from photocatalytic water splitting?

Yuhao Liu, Jinhua Xiong, Shuiguang Luo, Ruowen Liang, Na Qin, Shijing Liang\* and Ling Wu\*

Ultrathin monolayer HNbWO<sub>6</sub> nanosheets have been successfully prepared through a simple and ultrafast ion intercalation assisted exfoliation method. These obtained highly dispersed nanosheets present enhanced photocatalytic hydrogen evolution activity compared to the nanosheets prepared by the traditionally time-consuming process.

The discovery of the physical exfoliation of graphite into merely one carbon atom thick graphene single sheets has triggered the exponential growth of research and development of this fascinating ultrathin nanomaterial.<sup>1</sup> These single sheets, so-called nanosheets, can be regarded as a new class of materials, which have an extremely small thickness of  $\sim 1$  nm and are up to several tens of micrometers in lateral size.<sup>2,3</sup> In particular, ultrathin metal oxide nanosheets have drawn tremendous attention owing to their various elemental composition and distinctive properties, such as high surface area, anisotropic features, and singlecrystalline texture. 4-6 The unique characteristics of these nanostructures make them functional materials in fields including cosmetics,7 optoelectronics8 and heterogeneous photocatalysis.9-12 More importantly, the single molecular layers of these materials provide an opportunity for the understanding of the photocatalytic reaction at the molecular level. 13,14

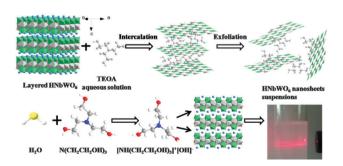
Generally, the effective methods of approach to the synthesis of two-dimensional (2D) nanosheets can be divided into two major categories: "top-down" and "bottom-up" methods according to the process of creating nanoscale structures. 15-18 Until now, several ultrathin metal oxide nanostructures have been prepared using a bottom-up approach through either the vapor deposition method<sup>19</sup> or wet-chemical one-pot synthesis, 20,21 but this approach needs the molecular or atomic components to build more complex nanoscale assemblies or directed self-assemblies based on complex mechanisms and technologies. 22,23 Therefore, ultrathin nanosheets are mainly prepared using a much milder top-down approach

State Key Laboratory of Photocatalysis on Energy and Environment, Fuzhou University, Fuzhou 350002, P. R. China. E-mail: wuling@fzu.edu.cn, sjliang2011@gmail.com; Fax: +86-591-83779105, +86-591-22866070; Tel: +86-591-83779362, +86-591-18060611675

through ion-exchange, intercalation, swelling, mechanical shaking and subsequent reassembly processes.<sup>24,25</sup> However, this type of exfoliation is considered to be time-consuming owing to slow diffusion processes and takes typically 1-2 weeks according to several reports.<sup>26</sup> In view of this, it is necessary to develop a very rapid and time-efficient approach for the preparation of 2D metal oxide nanosheets.

Herein, we report an efficient, simple method towards the ultrafast synthesis of ultrathin metal oxide nanosheets based on an acid-base reaction and ion intercalation assisted exfoliation strategy. Simply by dispersing the layered parent of HNbWO6. 1.5H<sub>2</sub>O in the polar amine of triethanolamine aqueous solution, translucent colloidal nanosheet suspensions can be obtained in just a few minutes. Scheme 1 illustrates the procedure for the fabrication of 2D ultrathin HNbWO6 nanosheets. It is found that thus obtained well-dispersed ultrathin monolayer HNbWO<sub>6</sub> nanosheet suspensions exhibit an efficient photocatalytic H<sub>2</sub> evolution performance via an in situ reaction under simulated sunlight irradiation. Moreover, highly dispersed and hyperfine Pt nanoparticle loading on the ultrathin HNbWO<sub>6</sub> nanosheets via an in situ photodeposition method showed a remarkable promotion for photocatalytic hydrogen production.

Monolayer HNbWO<sub>6</sub> nanosheet suspensions were synthesized rapidly using triethanolamine (TEOA) (see the ESI† for details).



Scheme 1 Scheme illustrating the preparation of ultrathin HNbWO<sub>6</sub> nanosheets through the acid-base reaction and ion intercalation assisted exfoliation strategy

<sup>†</sup> Electronic supplementary information (ESI) available. See DOI: 10.1039/c5cc05788d

Communication ChemComm

The samples are denoted as N-HNbWO<sub>6</sub>. Restacked HNbWO<sub>6</sub> nanosheet aggregates synthesized using the traditional approach with tetra-n-butylammonium hydroxide (TBAOH) are denoted as T-HNbWO<sub>6</sub>. Structural analysis of the synthetic samples was carried out by XRD (Fig. S1, ESI†). The single-phase diffraction pattern of layered LiNbWO<sub>6</sub> and HNbWO<sub>6</sub> samples matches well with the published data (JCPDS: 41-0378 and 41-0110, respectively). The characteristic (002) peak shifted to a lower  $2\theta$  angle after the proton-exchange process, indicating that the interlayer distance of  $HNbWO_6$  (d = 1.30 nm) is wider than that of LiNbWO<sub>6</sub> (d = 0.93 nm). This will be beneficial for the intercalation of an organic base in a subsequent exfoliation process. Compared with the layered parent compounds, HNbWO<sub>6</sub> nanosheets obtained using this rapid acid-base reaction method with TEOA aqueous solution lost their periodic layered structure and long-range order characteristics in c axial direction, only showing a few weak in-plane diffraction peaks of (110), (200) and (220). Owing to the disordered restacking, the (002) peak further shifted to a lower  $2\theta$  angle at  $6.34^{\circ}$ . These results are consistent with the T-HNbWO<sub>6</sub> nanosheets.

A typical layer structure of LiNbWO6 and HNbWO6 was observed to be stacked layer by layer from the SEM images shown in Fig. S2a (ESI†) and Fig. 2b, respectively. These layered metal oxides consist of tabular particles with a size of 3-20 μm, while the thickness of layered HNbWO<sub>6</sub> flakes is about 50 nm. The image of the HNbWO<sub>6</sub> nanosheets prepared by this rapid acid-base reaction method indicates the formation of a random aggregation of rather loose and irregular 2D nanosheets (Fig. S2c, ESI†).<sup>27</sup> This image is similar to that of the nanosheets obtained using the traditional method (Fig. S2d, ESI†). The morphology of these rapidly synthesized nanosheets was further confirmed by TEM and HRTEM images (Fig. 1). The individual sheets are obtained using this approach (Fig. 1a), with edge lengths of several hundred nanometers. Note that at the edge of the nanosheets, nanotubes were formed by the rolling up of exfoliated HNbWO<sub>6</sub> nanosheets. The produced nanotubes were usually 40-50 nm in diameter and several hundred nanometers in length.<sup>28</sup> The difference of chemical exfoliation reagents caused the cooperative distortion of MO<sub>6</sub> octahedra in HNbWO<sub>6</sub>, which might become a source of internal stress and induce the rolling-up behavior.<sup>29,30</sup> The HNbWO<sub>6</sub> nanosheet suspensions obtained using this method has an obvious Tyndall effect shown in Fig. 1a, inset, indicating the formation of freestanding and homogeneous 2D ultrathin nanosheets. As shown in Fig. 1b, the SAED analysis of an individual nanosheet reveals some bright spots with the single crystalline feature, which can be indexed as the [001] zone axis of the tetragonal lattice in planar orientation. The HRTEM image (Fig. 1c) shows a clear lattice structure, which demonstrates that the HNbWO6 nanosheets retain a high in-plane crystallinity. The angle labeled in the enlarged HRTEM image (Fig. 1d) is 90°, which is in agreement with the theoretical value of the angle between the (010) and (100) planes. The clear fringes have a lattice spacing of 0.33 nm, which corresponds to the (110) plane of tetragonal HNbWO<sub>6</sub>. Visualization of the nanosheets can be realized by atomic force microscopy (AFM) shown in Fig. 1e, exfoliated nanosheets with a lateral size in the range of submicrometers, and most of the measured

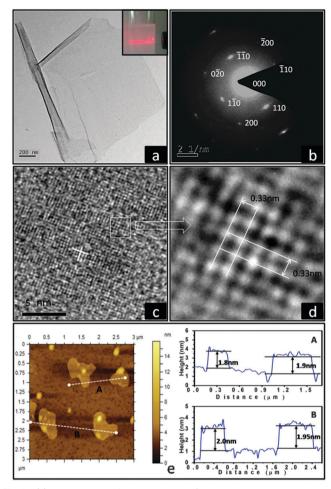


Fig. 1 (a) TEM images of ultrathin HNbWO<sub>6</sub> nanosheet suspensions. The inset in (a) is the photograph of the colloidal HNbWO<sub>6</sub> nanosheet dispersions. The light beam is incident from the side to demonstrate the Tyndall effect. (b) SAED and (c) HRTEM images taken from a part of a single nanosheet. (d) HRTEM image enlarged from (c). (e) Representative AFM image (left) and corresponding thickness analysis (right) taken around the white line in left.

thickness values of single layer HNbWO6 nanosheets are between 1.8 and 2.0 nm. This result well matched the expected value of the thickness of the HNbWO<sub>6</sub> layer (Fig. S3, ESI†), indicating that the as-prepared HNbWO6 nanosheets were monolayered. The difference between experimental height and slab thickness might arise from the distortion of the monolayer nanosheet and the adsorption of triethanolamine molecules and water molecules on the surface of the nanosheets (see Fig. S4, ESI†). Such a behavior was in accordance with previous results for oxide nanosheet systems. 31,32

In addition, energy dispersive X-ray spectrometry (EDS) analysis was conducted to confirm the composition of these HNbWO<sub>6</sub> nanosheets (Fig. S5a, ESI†). The elements, Nb, W and O, were all detected. The EDS mapping in Fig. S5b (ESI†) clearly shows the strong signals for Nb, W and O, further confirming that this method is an effective and fast method for preparing ultrathin nanosheets. In conclusion, the above mentioned results provide direct and solid evidence of the formation of monolayer HNbWO6 nanosheets. For comparison, the TEM and

ChemComm Communication

AFM of HNbWO<sub>6</sub> nanosheets prepared by the traditional method with TBAOH aqueous solution are also shown in Fig. S6a and b (ESI†), respectively. All these results indicate that this method is more advantageous than the traditional approach for the preparation of monolayer HNbWO6 nanosheets. The introduction of TEOA as a stripping agent based on an acid-base reaction and ion intercalation process is a key factor that promotes the exfoliation of the layered precursors. It is believed that the amine end of the TEOA tends to accept a proton from water because of the presence of lone electron pairs, generating [HN(CH<sub>2</sub>CH<sub>2</sub>OH)<sub>3</sub>]<sup>+</sup>[OH]<sup>-</sup>, which can passively diffuse between the two regions along with H<sup>+</sup> and OH<sup>-</sup>. In situ information of [HN(CH<sub>2</sub>CH<sub>2</sub>OH)<sub>3</sub>]<sup>+</sup>[OH]<sup>-</sup> between the neighboring layers offered this layered material for stripping. 33-35 We also demonstrated that this method was suitable to obtain HNbxTa1-xWO<sub>6</sub> (x = 0.7, 0.5,0.3, 0) ultrathin nanosheets (see Fig. S7, ESI†), suggesting the generality of the protocol. In addition, the stripping agent can also be replaced with other polar amines of triisopropanolamine (TIPA) for the efficient and ultrafast preparation of ultrathin HNbWO<sub>6</sub> nanosheets (see Fig. S8, ESI†).

The BET surface area of restacked HNbWO6 nanosheets prepared using TEOA and TBAOH aqueous solutions is about 41.4 and 84.5 m<sup>2</sup> g<sup>-1</sup>, respectively, which is larger than that of layered HNbWO<sub>6</sub> nanosheets (10.7 m<sup>2</sup> g<sup>-1</sup>). The adsorptiondesorption isotherm of restacked HNbWO<sub>6</sub> nanosheets is a type IV isotherm, whereas the layered HNbWO<sub>6</sub> displays the type II isotherm (Fig. S9a, ESI†). The pore size distribution of restacked nanosheets mainly ranges from 0.1 to 30 nm (Fig. S9b, ESI†), suggesting that the as-exfoliated HNbWO6 maintains the freestanding nanosheet form. The band gap energy and electronic properties of prepared nanosheets were calculated from UV-vis diffuse reflectance spectra and Mott-Schottky measurements, respectively. As shown in Fig. S10 (ESI†), the flat-band potential of N-HNbWO<sub>6</sub>, T-HNbWO<sub>6</sub> and L-HNbWO<sub>6</sub> was ca. -0.88 V, -0.84 V and -0.94 V vs. Ag/AgCl at pH 6.8, respectively, corresponding to ca. -0.68 V, -0.64 V and -0.74 V vs. NHE at pH 6.8, respectively, which is more negative than the H<sub>2</sub>O/H<sub>2</sub> potential (-0.41 V vs. NHE at pH 6.8). Therefore, it is thermodynamically permissible for the transformation of photogenerated electrons into H<sub>2</sub>O to produce H<sub>2</sub>.

Using these highly dispersed monolayer HNbWO<sub>6</sub> nanosheet suspensions as heterogeneous photocatalysts, H2 evolution reactions were conducted under simulated sunlight irradiation (Fig. S11, ESI†). For comparison, the photocatalytic performances of restacked HNbWO6 nanosheets obtained with TBAOH were further evaluated by redispersing them in the TEOA aqueous solution. As presented in Fig. 2, N-HNbWO<sub>6</sub> and T-HNbWO<sub>6</sub> show negligible activity for hydrogen evolution without Pt modification. In contrast, the activity of both improved remarkably with the Pt loading due to the Schottky barrier junctions. Moreover, the average hydrogen production rate of N-HNbWO6 is strikingly higher than T-HNbWO<sub>6</sub>. The reduction in the activity of T-HNbWO<sub>6</sub> could be attributed to a mild blockage of surface sites by the adsorption of bulkier TBA+ cations.26 The TEM of restacked HNbWO6 nanosheets (Fig. S12, ESI†) also further confirmed that these aggregates were comprised of irregularly

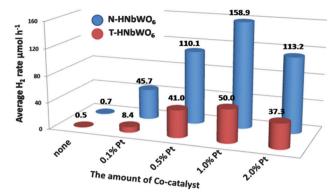


Fig. 2 Average H<sub>2</sub> evolution rate of N-HNbWO<sub>6</sub> and T-HNbWO<sub>6</sub> loaded with different amounts of the Pt co-catalyst reaction for five hours.

stacked and folded nanosheets, and no single sheets were observed. Furthermore, the hydrogen production amount of 1.0 wt% Pt loaded N-HNbWO $_6$  reached up to 794.5  $\mu$ mol for five hours (with a stable H $_2$  evolution rate of 158.9  $\mu$ mol h $^{-1}$ ), which was much higher than that of T-HNbWO $_6$  (Fig. S13, ESI $^{\dagger}$ ). The apparent quantum yield of 1.0 wt% Pt loaded N-HNbWO $_6$  and T-HNbWO $_6$  was 14.1% and 2.4%, respectively (Fig. S14, ESI $^{\dagger}$ ).

In heterogeneous photocatalysis, the size and uniformity of a co-catalyst have a significant impact on the photocatalytic water splitting performance, and the smaller size and high dispersion of cocatalyst particles are more active for H<sub>2</sub> evolution.<sup>36,37</sup> From Fig. 3a and b, the Pt nanoparticles anchored on these nanosheet suspensions were smaller and more dispersed, while some Pt particle agglomerates were formed on the restacked T-HNbWO<sub>6</sub> nanosheets. Thus, the enhancement of the photocatalytic activity of the well-dispersed ultrathin monolayer nanosheet suspensions can be attributed to much higher dispersion and smaller size of Pt nanoparticles anchored onto their surfaces. ICP results show that the percentage of Pt in the samples N-HNbWO<sub>6</sub> and T-HNbWO6 was 0.88% and 0.92%, respectively, further revealing that the difference in the activities between these two samples was not caused by the different contents of Pt loading. All these results indicate that highly dispersed and hyperfine Pt nanoparticles were key factors for the enhanced photocatalytic H<sub>2</sub> production activity of N-HNbWO<sub>6</sub>.

In summary, ultrathin HNbWO<sub>6</sub> nanosheets have been successfully obtained *via* an ultrafast, simple and efficient

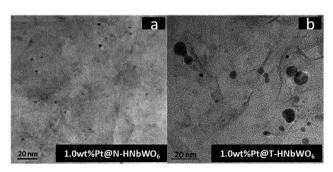


Fig. 3 TEM images of 1.0 wt% Pt loaded N-HNbWO $_6$  (a) and T-HNbWO $_6$  nanosheets (b).

Communication ChemComm

exfoliation method based on an acid-base reaction and ion intercalation strategy. The as-prepared HNbWO<sub>6</sub> ultrathin nanosheet suspensions are stable and highly dispersed. Compared with the HNbWO<sub>6</sub> nanosheet aggregates prepared using TBAOH aqueous solution, the ultrathin HNbWO6 nanosheet suspensions showed efficient activity for photocatalytic H<sub>2</sub> evolution, due to the in situ deposition of smaller and highly dispersed noble-metal clusters. We believe that our strategy could be broadly applied for the facile production of other ultrathin nanostructures with great potential for use in various applications.

This work was supported by the National Natural Science Foundation of China (21177024, 21303019 and 21273036) and the Natural Science Foundation of Fujian Province (2014J05016).

## Notes and references

- 1 K. S. Novoselov, A. K. Geim, S. Morozov, D. Jiang, Y. Zhang, S. A. Dubonos, I. Grigorieva and A. Firsov, Science, 2004, 306, 666-669.
- 2 T. Shibata, G. Takanashi, T. Nakamura, K. Fukuda, Y. Ebina and T. Sasaki, Energy Environ. Sci., 2011, 4, 535-542.
- 3 L. Wang and T. Sasaki, Chem. Rev., 2014, 114, 9455-9486.
- 4 S. I. Shin, A. Go, I. Y. Kim, J. M. Lee, Y. Lee and S.-J. Hwang, Energy Environ. Sci., 2013, 6, 608-617.
- 5 M. Xu, T. Liang, M. Shi and H. Chen, Chem. Rev., 2013, 113, 3766-3798.
- 6 S. Z. Butler, S. M. Hollen, L. Cao, Y. Cui, J. A. Gupta, H. R. Gutierrez, T. F. Heinz, S. S. Hong, J. Huang and A. F. Ismach, ACS Nano, 2013, 7, 2898-2926.
- 7 A. López-Galindo, C. Viseras and P. Cerezo, Appl. Clay Sci., 2007, 36, 51-63.
- 8 M. Osada and T. Sasaki, Adv. Mater., 2012, 24, 210-228.
- 9 T. W. Kim, S. G. Hur, S.-J. Hwang, H. Park, W. Choi and J.-H. Choy, Adv. Funct. Mater., 2007, 17, 307.
- 10 J. L. Gunjakar, T. W. Kim, H. N. Kim, I. Y. Kim and S.-J. Hwang, J. Am. Chem. Soc., 2011, 133, 14998-15007.
- 11 K. Maeda, M. Eguchi, W. J. Youngblood and T. E. Mallouk, Chem. Mater., 2009, 21, 3611-3617.
- Y. Okamoto, S. Ida, J. Hyodo, H. Hagiwara and T. Ishihara, J. Am. Chem. Soc., 2011, 133, 18034-18037.
- 13 J. Meurigá Thomas, Chem. Commun., 2006, 3273-3278.

- 14 S. Liang, L. Wen, S. Lin, J. Bi, P. Feng, X. Fu and L. Wu, Angew. Chem., 2014, 126, 2995-2999.
- 15 T. Sasaki and M. Watanabe, J. Am. Chem. Soc., 1998, 120, 4682-4689.
- 16 F. Geng, R. Ma, Y. Yamauchi and T. Sasaki, Chem. Commun., 2014, 50, 9977-9980,
- 17 M. A. Bizeto, A. L. Shiguihara and V. R. Constantino, J. Am. Chem. Soc., 2009, 19, 2512-2525.
- 18 Z. Zhai, C. Hu, X. Yang, L. Zhang, C. Liu, Y. Fan and W. Hou, J. Mater. Chem., 2012, 22, 19122–19131.
- 19 T. Orzali, M. Casarin, G. Granozzi, M. Sambi and A. Vittadini, Phys. Rev. Lett., 2006, 97, 156101.
- 20 E. L. Tae, K. E. Lee, J. S. Jeong and K. B. Yoon, J. Am. Chem. Soc., 2008, 130, 6534-6543.
- 21 S. Liang, S. Zhu, Y. Chen, W. Wu, X. Wang and L. Wu, J. Mater. Chem., 2012, 22, 2670-2678.
- 22 K. Sakakibara, J. P. Hill and K. Ariga, Small, 2011, 7, 1288-1308.
- 23 S. Acharya, J. P. Hill and K. Ariga, Adv. Mater., 2009, 21, 2959–2981. 24 T. Sasaki, M. Watanabe, H. Hashizume, H. Yamada and H. Nakazawa, Chem. Commun., 1996, 229-230.
- 25 M. Leng, Y. Chen and J. Xue, Nanoscale, 2014, 6, 8531–8534.
- 26 K. Nakagawa, T. Jia, W. Zheng, S. M. Fairclough, M. Katoh, S. Sugiyama and S. C. E. Tsang, Chem. Commun., 2014, 50, 13702-13705.
- 27 C. Hu, L. Zhang, L. Cheng, J. Chen, W. Hou and W. Ding, J. Energy Chem., 2014, 23, 136-144.
- 28 P. Liu, H. Zhang, H. Liu, Y. Wang, X. Yao, G. Zhu, S. Zhang and H. Zhao, J. Am. Chem. Soc., 2011, 133, 19032-19035.
- 29 R. E. Schaak and T. E. Mallouk, Chem. Mater., 2000, 12, 3427-3434. 30 G. B. Saupe, C. C. Waraksa, H.-N. Kim, Y. J. Han, D. M. Kaschak,
- D. M. Skinner and T. E. Mallouk, Chem. Mater., 2000, 12, 1556–1562. 31 P. Shen, H. Zhang, H. Liu, J. Xin, L. Fei, X. Luo, R. Ma and S. Zhang, J. Mater. Chem. A, 2015, 3, 3456-3464.
- 32 K. Akatsuka, G. Takanashi, Y. Ebina, M.-a. Haga and T. Sasaki, J. Phys. Chem. C, 2012, 116, 12426-12433.
- 33 F. Geng, R. Ma, A. Nakamura, K. Akatsuka, Y. Ebina, Y. Yamauchi, N. Miyamoto, Y. Tateyama and T. Sasaki, Nat. Commun., 2013, 4, 1632.
- 34 F. Geng, R. Ma, Y. Ebina, Y. Yamauchi, N. Miyamoto and T. Sasaki, J. Am. Chem. Soc., 2014, 136, 5491-5500.
- 35 H. Yuan, D. Dubbink, R. Besselink and J. E. ten Elshof, Angew. Chem., Int. Ed., 2015, 54, 9239-9243, DOI: 10.1002/anie.201502539.
- 36 E. M. Sabio, R. L. Chamousis, N. D. Browning and F. E. Osterloh, J. Phys. Chem. C, 2012, 116, 3161-3170.
- 37 T. Oshima, D. Lu, O. Ishitani and K. Maeda, Angew. Chem., Int. Ed., 2015, 54, 2698-2702.