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Sponge-like NiCo₂O₄/MnO₂ ultrathin nanoflakes for supercapacitor with high-rate performance and ultra-long cycle life†

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We present a simple strategy for synthesizing sponge-like $NiCo_2O_4/MnO_2$ ultrathin nanoflakes, which exhibit a high specific capacitance of 935 F g⁻¹ at 1 A g⁻¹, excellent rate performance (74.9% retention at 50 A g⁻¹), and ultra-long cycling stability (103.1% of the initial capacitance after 25 000 cycles).

Transition metal oxides (TMOs), especially in the form of nanomaterials, had been extensively studied as electrode materials of pseudocapacitors in the past decades. 1-5 This can be interpreted as their abundant Faradaic redox reactions, which contribute to high specific capacitances.6 However, TMOs with which redox reactions are involved often suffer from a lack of stability during long-term cycling.7 Furthermore, the low electronic conductivity of TMOs profoundly restricts their rate performance.8,9 An attractive concept to alleviate the previously mentioned negative effects is to combine two or more types of TMOs. Mixed TMOs as prospective electrode materials, which could inherit advantages from individual TMOs and generate excellent properties, have received an upsurge of interest in recent years. For instance, Xia et al.10 reported the Co3O4/NiO core-shell nanowire arrays for pseudocapacitors with an enhanced capacitance (853 F g⁻¹ at 2 A g⁻¹), rate performance and electrochemical stability; Liu et al.¹¹ reported the MnO₂/NiO tube arrays for pseudocapacitors with an improved capacitance (0.35 F cm⁻² at 8.5 mA cm⁻²) with respect to pristine MnO2 tubular arrays, a good cycle performance, and remarkable rate capability.

In addition to anticipating composition, the performance of TMOs in various applications also heavily depends on the

Based on the previous discussion, we present a simple strategy to fabricate sponge-like NiCo2O4/MnO2 ultrathin nanoflakes on graphite paper with remarkable capacitive performance. NiCo2O4 and MnO2 are composited into spongelike ultrathin nanoflakes via a simple one-step electrodeposition with subsequent annealing. When applied as an electrode, the smartly designed composites possess three merits: (I) ultrathin nanoflakes would enable a fast, reversible Faradaic reaction and provide a short ion diffusion path, hence a good rate performance can be obtained;15 (II) the fluffy, sponge-like structure is advantageous to the permeation of the electrolyte to the inner electrode, as well as the emission of joule heat to the electrolyte, thus increasing the thermal stability of the structure;16,17 and (III) ultrathin nanoflakes are strongly supported on graphite paper, in which binding and conductive additives can be avoided and, consequently, the inactive surface is significantly reduced.18 The electrochemical measurements show that the specific capacitance of the sponge-like NiCo₂O₄/MnO₂ ultrathin nanoflakes achieves a maximum of 935 F g^{-1} at a current density of 1 A g⁻¹, which could still retain 700 F g⁻¹ (74.9% retention) even if the current density reaches as high as 50 A g⁻¹. Moreover, the electrode maintains 103.1% capacitance of initialization after 25 000 cycles, illustrating ultra-long cycling stability. These results show that the sponge-like NiCo₂O₄/MnO₂ ultrathin nanoflakes should be a very promising electrode material for supercapacitors.

microtexture of the materials. ¹² Therefore, the rational design of electrode materials with well-defined micro/nanostructures is imperative for further enhancement of the electrochemical properties. Hierarchical structures, which draw special interest due to their exceptional properties and remarkable potential in many fields, have been synthesized for many simple metal oxides, but the facile synthesis of the mixed TMOs with desirable composition and hierarchical structure still remains a challenge. ^{13,14} Thus, it is desired that well-designed compositions for supercapacitors with encouraging electrochemical performance, especially that of high-rate performance and long-term cycling stability, could be synthesized by a facile method.

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Fig. 1a shows the X-ray diffraction (XRD) pattern of the NiCo2O4/MnO2 ultrathin nanoflakes. As observed in Fig. 1a, except for the two strong peaks from the graphite paper substrate, other well-defined diffraction peaks confirm the existence of the spinel NiCo₂O₄ phase (JCPDS card no. 20-0781). The peak at 12.9° comes from the tetragonal MnO₂ phase (JCPDS card no. 44-0141). The chemical composition of the sample was further analyzed by X-ray photoelectron spectroscopy (XPS) shown in Fig. S1 (ESI†). High-resolution XPS spectra of Mn 2p, Co 2p and Ni 2p for the NiCo2O4/MnO2 ultrathin nanoflakes are presented in Fig. 1b-d. By using a Gaussian fitting method, Fig. 1b of the Mn 2p region demonstrates two main peaks at 642.4 and 653.7 eV, indicating that the elemental Mn is in the formation of MnO2.11,19,20 The Co 2p spectrum (Fig. 1c) is best fitted considering the two spin-orbit doublets characteristic of Co2+ and Co3+ and one shakeup satellite (identified as "Sat."). The Ni 2p spectrum (Fig. 1d) is best fitted considering the two spin-orbit doublets characteristic of Ni²⁺ and Ni³⁺ and two shakeup satellites. These results show that the NiCo₂O₄/MnO₂ ultrathin nanoflakes have a composition containing Co2+, Co3+, Ni2+ and Ni3+, which is in good agreement with literature for NiCo₂O₄. 18,21 Additionally, the uniform distribution of Mn, Ni and Co species is confirmed by the linescan mapping (Fig. S2, see ESI†), which reveals that the composition is mainly composed of NiCo2O4 and only contains a few of the MnO₂.

Low- and high-magnification field emission scanning electron microscopy (FESEM) images of the sample are shown in Fig. 2. A low FESEM image viewed from the top can be seen in Fig. 2a, which indicates that the entire material on the graphite paper is homogeneous. A high-magnification image of the sample is shown in Fig. 2b, which reveals that the architecture is composed of numerous ultrathin nanoflakes (<10 nm) and has a cross-linked character, in which channels and pores are formed. Low- and high-magnification FESEM images of the cross section are available in Fig. 2c and d, respectively. Fig. 2c reveals that the thickness of the architecture is \sim 2 μ m. Fig 2d

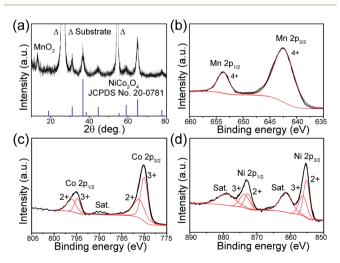


Fig. 1 (a) XRD pattern of the NiCo₂O₄/MnO₂ ultrathin nanoflakes. (b) Mn 2p, (c) Co 2p and (d) Ni 2p spectra of the $NiCo_2O_4/MnO_2$ ultrathin nanoflakes

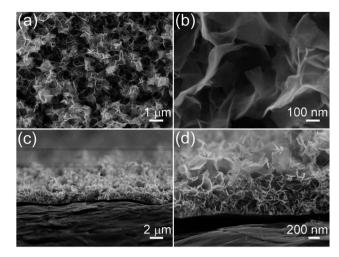


Fig. 2 (a) Low- and (b) high-magnification FESEM images from the top view, and (c) low- and (d) high-magnification FESEM images of the side

elucidates that the ultrathin nanoflakes are standing over the graphite paper; thus, it is beneficial to the charge transport and ion diffusion without the necessity of binder blocks, which results in improved charge transfer kinetics. Notably, the crosslinked ultrathin nanoflakes provide a fluffy, sponge-like architecture to facilitate the permeation of the electrolyte to the inner electrode, as well as the emission of joule heat to the electrolyte.

To better understand the structure, representative transmission electron microscopy (TEM) images taken from the sample are given in Fig. 3a and b. The dark strips in Fig. 3a are generally the folded edges or wrinkles of the nanoflakes. The magnified images of Fig. 3b clearly show that mesopores are uniformly distributed throughout the entire surface of the nanoflakes. The sizes of the mesopores are estimated to be in the range of 2-5 nm. It is well known that the mesoporous structures in nanoflakes facilitate the mass transport of electrodes within

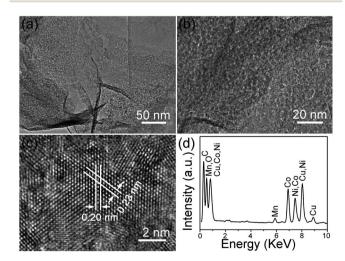


Fig. 3 (a) Low-magnification TEM image. (b) Enlarged TEM image. (c) HRTEM image of NiCo₂O₄/MnO₂ ultrathin nanoflakes. (d) EDX pattern taken from ultrathin nanoflakes.

electrolytes for Faradaic redox reactions.^{7,22} The mesoporous structure will also enlarge the electrode/electrolyte contact areas, which further enhances the electrochemical performance of electrode.¹² As can be seen from the HRTEM image of Fig. 3c, the lattice fringes give an interplanar spacing of 0.23 nm and 0.20 nm, corresponding to that of the {222} and {400} lattice planes of the spinel NiCo₂O₄ crystal, respectively. The EDX spectrum is shown in Fig. 3d. The presence of Mn, Co and Ni is in agreement with our designed concept, and the Cu and C signals clearly come from the TEM grid.

The electrochemical measurements were performed on an Autolab (PGSTAT302N) with a three-electrode configuration system in 1 M KOH aqueous solution within the potential window of -0.1 to 0.5 V (vs. Ag/AgCl) to evaluate the electrochemical performance of the as-synthesized electrodes. The samples vary according to the electrodeposition time, particularly 20, 30 and 40 min (denoted as P1, P2 and P3, respectively). The specific capacitance values of the samples had been determined by cyclic voltammetry (CV) and galvanostatic charge-discharge (CD) measurements. Fig. 4a shows the CV curves of the P1 sample at different scan rates. Strong redox peaks over the entire range of scan rates can be clearly observed, which reveals that the capacitive characteristics are mainly governed by the Faradaic reaction. It is notable that the capacitance contributed from the graphite paper substrate is negligible (Fig. S3, ESI†).

It is well accepted that galvanostatic charge-discharge examination is an established method for estimating specific

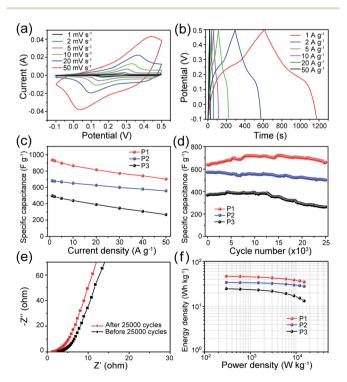


Fig. 4 (a) CV and (b) CD curves of the P1 sample. (c) Specific capacitance as a function of the current of P1, P2 and P3 samples. (d) Cycling performances of P1, P2 and P3 samples at a scan rate of 50 mV s $^{-1}$. (e) Nyquist plot of the P1 sample. (f) Ragone plots of P1, P2 and P3 samples.

capacitance. Fig. 4b presents the typical charge-discharge voltage vs. the time plots of the P1 sample at various current densities. A symmetric nonlinear triangular shape is observed, further indicating their pseudo-capacitance behavior and excellent reversibility. The specific capacitance of the P1 sample at various current densities can be calculated based on the corresponding CD curves, and the results are depicted in Fig. 4c together with the other two samples. The P1 sample manifests high specific capacitance values of 935, 923, 895, 863, 813, 770, 740 and 700 F g^{-1} at current densities of 1, 2, 5, 10, 20, 30, 40 and 50 A g⁻¹, respectively, which is superior to the other two samples (Table S1 in ESI†). This is mainly due to that as for the P2 and the P3 sample, a dense layer (Fig. S4, ESI†) covering on the sponge-like ultrathin nanoflakes prevents the electrolyte from permeating into the sponge-like ultrathin nanoflakes, and only the outer surface of the dense layer can be utilized for the charge storage. It can be deduced that the thickness of the dense layer should increase as electrodeposition time goes on, and then specific capacitance values would decrease. The CD curves of the P2 and the P3 sample are shown in Fig. S5a and b (ESI†) for further demonstration. All of these samples simultaneously show high rate capabilities (74.9% for P1, 81.8% for P2, and 53.7% for P3) when current densities increase from 1 to 50 Ag^{-1} , which can be ascribed to that ultrathin nanoflakes enable a fast Faradaic reaction and provide a short ion diffusion path.

Cycling stability is another critical factor in evaluating the electrochemical properties of supercapacitors. The cycling stability curves of the three electrodes at a scan rate of 50 mV s⁻¹ are illustrated in Fig 4d. The final specific capacitance of the P1 sample retained 103.1% of its initial value when it had undergone 25 000 cycles. In other words, the loss in specific capacitance based on the maximum value is only ~8.7%. Such an ultralong cycle life that can be ascribed to the fluffy sponge-like structure is advantageous to the emission of joule heat to the electrolyte, hence increasing the thermal stability of the structure. However, cycling performances of the P2 (87.8% retention) and P3 (71.1% retention) samples are slightly satisfactory. It can be attributed to the denser layers of the P2 and P3 samples, which cannot adapt well to the pronounced volume expansion/ contraction and result in the pulverization of the electrode material. The Nyquist plots of the P1 sample are also presented in Fig. 4e. It is known that the equivalent series resistance (ESR) is an important characteristic of supercapacitors measured at highfrequency regions where the curve intersects on the real axis. The higher ESR indicates the lower electrical conductivity of the sample. It can be seen that the P1 sample shows ESR value \sim 2.7 Ω before cycling. After 25 000 cycles, the P1 sample exhibits lower ESR (\sim 1.3 Ω), which is due to the activation of active material during the cycling process. In addition, after 25 000 cycles, the slope is close to 90° at the low-frequency region, which indicates that the P1 sample has good capacitive behaviour.

In view of the excellent rate capability and ultra-long cycling life, we further calculate the energy and power densities, which are the two key factors for the practical applications of supercapacitors. The results are presented as Ragone plots in Fig. 4f. The P1 sample can deliver high power output in the range of 0.3 and 15 kW ${\rm kg}^{-1}$ with little sacrifice of the high energy density

(from 46.75 to 35 W h kg^{-1}), better than that of the P2 (34.1 to 27.9 W h kg^{-1}) and P3 (24.8 to 13.3 W h kg⁻¹) samples. This result further confirms that the sponge-like NiCo2O4/MnO2 ultrathin nanoflakes of the P1 sample is an excellent capacitive material.

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Based on previous discussions, we have determined that the sponge-like NiCo₂O₄/MnO₂ ultrathin nanoflakes of the P1 sample possessed the best electrochemical performance. However, it should be truly understood whether this strategy to synthesize NiCo2O4/MnO2 is of any advantage with respect to pure NiCo₂O₄ because NiCo₂O₄ exhibits good pseudocapacitive behavior.23-27 The pure NiCo2O4 was prepared by using the solution without Mn(NO₃)₂ under the same conditions. The electrochemical performances of the obtained pure NiCo₂O₄ and NiCo₂O₄/MnO₂ are shown in Fig. S6 (ESI†). Fig. S6a† shows the cyclic voltammetry (CV) curves of NiCo₂O₄/MnO₂ and pure NiCo₂O₄ at a scan rate of 10 mV s⁻¹. The specific capacitance of pure NiCo₂O₄ is 1005 F g⁻¹, which is higher than that of NiCo₂O₄/MnO₂ (839 F g⁻¹). Fig. S6b† shows the galvanostatic charge-discharge (CD) curves of NiCo2O4/MnO2 and pure $NiCo_2O_4$ at a current density of 5 A g^{-1} . The specific capacitance of pure NiCo₂O₄ calculated from CD curves is larger than that of NiCo₂O₄/MnO₂. However, when we focus on the rate performance and cycling stability of the materials, which are also as important as the specific capacitance, we determine that NiCo₂O₄/MnO₂ acquired high-rate performance and improved cycling stability than pure NiCo₂O₄. Fig. S6c† shows that NiCo₂O₄/MnO₂ exhibits a high-rate performance of 74.9% retention, compared with 70.3% retention from pure NiCo₂O₄. Fig. S6d† indicates that NiCo₂O₄/MnO₂ possesses ultra-long cycling stability (103.1% of the initial capacitance after 25 000 cycles), while pure NiCo₂O₄ can only retain 73.6% of its initial capacitance after 15 000 cycles. The excellent rate performance and ultra-long cycling stability of NiCo₂O₄/MnO₂ demonstrate that this strategy is advantageous.

In summary, sponge-like NiCo₂O₄/MnO₂ ultrathin nanoflakes on graphite paper were synthesized by a simple process of electrodeposition with subsequent annealing. The as-synthesized electrode by 20 min electrodeposition exhibited an enhanced specific capacitance of 935 F g⁻¹ at 1 A g⁻¹ with high rate performance (74.9% capacity retention at 50 A g⁻¹) and ultra-long cycling stability (maintaining 103.1% capacitance of initialization after 25 000 cycles), which make the sponge-like NiCo₂O₄/MnO₂ ultrathin nanoflakes a promising electrode material for high-performance supercapacitors.

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