

Low Temperature H<sub>2</sub>O and NO<sub>2</sub> Coadsorption on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) Ultrathin Films

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The coadsorption of H<sub>2</sub>O and NO<sub>2</sub> molecules on a well-ordered, ultrathin  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) film surface was studied using temperature programmed desorption (TPD), infrared reflection absorption spectroscopy (IRAS), and X-ray photoelectron spectroscopy (XPS). For H<sub>2</sub>O and NO<sub>2</sub> monolayers adsorbed separately on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface, adsorption energies were estimated to be 44.8 and 36.6 kJ/mol, respectively. Coadsorption systems prepared by sequential deposition of NO<sub>2</sub> and H<sub>2</sub>O revealed the existence of coverage and temperature-dependent adsorption regimes where H<sub>2</sub>O molecules and the surface NO<sub>x</sub> species (NO<sub>2</sub>/N<sub>2</sub>O<sub>4</sub>/NO<sub>2</sub><sup>−</sup>, NO<sub>3</sub><sup>−</sup>) form segregated and/or mixed domains. Influence of the changes in the crystallinity of solid water (amorphous vs crystalline) on the coadsorption properties of the NO<sub>2</sub>/H<sub>2</sub>O/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) system is also discussed.

## 1. Introduction

Interaction between condensed H<sub>2</sub>O and NO<sub>2</sub> layers on solid surfaces has a number of significant implications for various technological and environmental applications. Treatment of NO<sub>x</sub> emissions in automotive emissions using Pt/Rh/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>- or Pd/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>-based three-way catalysts;<sup>1–2</sup> NO<sub>x</sub> abatement of diesel engine emissions via Pt/BaO/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>-based NO<sub>x</sub> storage reduction (NSR) catalysts<sup>3</sup>, and photochemically induced atmospheric NO<sub>x</sub> reactions occurring on solid H<sub>2</sub>O surfaces in the stratosphere leading to the formation of acid rain and the depletion of the ozone layer<sup>4</sup> can be considered as some of the important examples where interaction of NO<sub>2</sub> and H<sub>2</sub>O layers plays a key role.

For this purpose, well-defined model catalyst systems serve as a convenient platform to study NO<sub>2</sub>, H<sub>2</sub>O, and NO<sub>2</sub>+H<sub>2</sub>O layers at low temperatures. A majority of the previous surface science studies on model catalysts were performed on metallic surfaces such as Au(111),<sup>5–9</sup> polycrystalline Au,<sup>10</sup> polycrystalline Cu,<sup>11</sup> polycrystalline Ni,<sup>12</sup> W(110),<sup>13</sup> Pt(111),<sup>14</sup> Pt(100),<sup>15</sup> Sn/Pt(111),<sup>16</sup> Rh(111)–Rh/Pd(111),<sup>17</sup> Ag(111),<sup>18–23</sup> and Ag(110).<sup>24</sup> In contrast to the metallic systems, there exists only a limited number of surface science studies focusing on the interaction of adsorbed NO<sub>2</sub> + H<sub>2</sub>O layers on metal oxide surfaces in the literature.<sup>25,26</sup> Due to its relevance to NSR catalysts, we have previously investigated H<sub>2</sub>O and NO<sub>2</sub> adsorptions on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) model catalyst support surfaces<sup>27,28</sup> as well as on BaO/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and Pt/BaO/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> high surface area catalysts.<sup>29–31</sup> Our previous results indicate that H<sub>2</sub>O adsorption on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface is predominantly molecular rather than dissociative. For  $\theta_{\text{H}_2\text{O}} < 1$  ML (ML = monolayer), H<sub>2</sub>O molecules were found to populate Al<sup>3+</sup> cation sites to form isolated H<sub>2</sub>O species aligned in a row along the cation sites on the oxide surface with a repulsive interaction between them. For  $\theta_{\text{H}_2\text{O}} > 1$  ML, ice multilayers were observed to form, which then desorb during temperature programmed desorption (TPD) with zero-order kinetics.<sup>27</sup> For the NO<sub>2</sub>/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) adsorption system, in addition to reversible molecular NO<sub>2</sub> adsorption/desorption

phenomena, some of the adsorbed NO<sub>2</sub> is converted to strongly bound nitrites and nitrates that are stable on the model catalyst surface at temperatures as high as 300 and 650 K, respectively.<sup>28</sup> The stability of the NO<sub>x</sub> formed by exposing the  $\theta$ -Al<sub>2</sub>O<sub>3</sub> model catalyst to NO<sub>2</sub> increases in the order: NO<sub>2</sub> (physisorbed or N<sub>2</sub>O<sub>4</sub>) < NO<sub>2</sub> (chemisorbed) < NO<sub>2</sub><sup>−</sup> < NO<sub>3</sub><sup>−</sup>.<sup>28</sup> In metal-oxide or zeolite based catalytic systems, it is known that adsorbed H<sub>2</sub>O can compete with various NO<sub>x</sub> species on the catalyst surface by occupying/transforming available adsorption sites and, hence, result in the modification of the catalytic NO<sub>x</sub> chemistry.<sup>3,32,33</sup> Therefore, in this work, we examine the NO<sub>2</sub> + H<sub>2</sub>O interactions on a  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) ultrathin film surface at low temperatures using TPD, infrared reflection absorption spectroscopy (IRAS), and X-ray photoelectron spectroscopy (XPS) techniques. First we will discuss the basic aspects of NO<sub>2</sub> and H<sub>2</sub>O adsorptions on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) model catalyst surface in a comparative fashion based on the results of TPD experiments. Next, the nature of the physisorbed NO<sub>2</sub>-multilayers and their surface configuration on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) is elucidated in the presence and absence of H<sub>2</sub>O via the IRAS technique. Finally NO<sub>2</sub> + H<sub>2</sub>O coadsorption by sequential deposition of H<sub>2</sub>O and NO<sub>2</sub> on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) is studied using TPD and XPS in order to investigate changes in the adsorption energies, adsorption sites, and oxidation states of the probe molecules, respectively.

## 2. Experimental Section

Experiments presented in this work were conducted in an ultrahigh vacuum (UHV) surface analysis chamber ( $P_{\text{base}} = 3 \times 10^{-10}$  Torr) equipped with XPS, Auger electron spectroscopy (AES), a quadrupole mass spectrometer (QMS) for TPD, and a rear-view low-energy electron diffraction (LEED) techniques. The main UHV chamber is coupled to a high-pressure IR cell ( $P_{\text{base}} \sim 2 \times 10^{-9}$  Torr) through a set of doubly differentially pumped sliding seals, which is equipped with CaF<sub>2</sub> windows. The NiAl(100) single crystal used in the experiments (Princeton Scientific Corp., 10 mm diameter, 2 mm thick) was polished on both sides and spot-welded onto a U-shaped Ta wire. A C-type thermocouple was spot welded to the top edge of the crystal for temperature measurements. The NiAl(100) crystal was cleaned by alternating cycles of Ar<sup>+</sup> ion sputtering ( $V_{\text{beam}}$

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= 1.5 kV,  $I_{\text{beam}} = 1.5 \mu\text{A}$ ) and high-temperature UHV anneals at 1200 K. Cleanliness of the NiAl(100) sample was checked with Auger electron spectroscopy (AES) and low energy electron diffraction (LEED).

Ultrathin  $\theta\text{-Al}_2\text{O}_3$  films on clean NiAl(100) were grown by adopting a procedure that was originally suggested by Ibach and co-workers<sup>34,35</sup> which included saturation of the NiAl(100) surface with  $\text{O}_2$  at 300 K (total  $\text{O}_2$  exposure = 9000 L;  $1 \text{ L} = 1 \times 10^{-6} \text{ Torr}\cdot\text{s}^{-1}$ ) and annealing of the O-saturated surface at 1200 K in UHV for 30–60 min in order to improve the crystallinity of the oxide film.<sup>36–38</sup> The quality of the  $\theta\text{-Al}_2\text{O}_3$  films were checked with AES, XPS, and LEED, and typical film thicknesses were  $6 \text{ \AA} \pm 2 \text{ \AA}$ .<sup>27</sup>

TPD experiments were carried out using a differentially pumped QMS (UTI) by applying  $-70 \text{ V}$  bias voltage on the spectrometer shield to constrain the ionizing electrons to the interior of the QMS shield to prevent any possible electron beam damage of the sample. All of the TPD data presented in this study were obtained by ramping the temperature of the crystal at a constant rate of  $2 \text{ K/s}$ . To minimize background desorption artifacts in the TPD data, a tubular pinhole doser, positioned in close proximity of the sample ( $\sim 2 \text{ mm}$  away), was used in the adsorption experiments, which allowed the background pressure in the chamber to stay in the  $\sim 10^{-10} \text{ Torr}$  range during the dosing processes.

Before the introduction of  $\text{NO}_2$  gas to the vacuum chamber, the gas dosing line and the pinhole doser were passivated by flushing  $\text{NO}_2$  gas through the dosing lines for an extended period of time. Passivation of the dosing line was also monitored with QMS by following the 46 amu signal with respect to the 30 amu signal for a constant flux of  $\text{NO}_2$ . The saturation of the 46 amu/30 amu ratio indicated the deactivation of the surfaces used in the dosing lines. This procedure was found to be helpful to minimize the decomposition of  $\text{NO}_2$  prior to the admittance to the vacuum chamber.

IRAS experiments were performed using a Mattson Research Series-1 FTIR spectrometer. For all of the IRAS data given here, the  $1000\text{--}4000 \text{ cm}^{-1}$  region was scanned for 16 min/spectrum (3420 scans/spec) with a resolution of  $4 \text{ cm}^{-1}$ . A background spectrum prior to each experimental series was collected from the clean  $\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$  surface at 80 K.

XPS data were acquired with a dual anode X-ray source and multichannel electrostatic hemispherical electron energy analyzer (Omicron, EA-125), using  $\text{AlK}\alpha$  X-rays ( $h\nu = 1486.6 \text{ eV}$ ) and a 50 eV analyzer pass energy. The X-ray source was oriented  $\sim 50^\circ$  with respect to the sample normal. XPS data were analyzed by fitting the minimum number of Gaussian peaks possible to the experimental spectra. Typical fwhm of 2.45–2.50 eV for N1s features and 2.50–2.55 eV for O1s features were used, and the resulting residual curves had a total integrated intensity of  $\leq 3\%$  of the corresponding experimental counterparts. Experimental XPS data were calibrated so that the  $\text{Al}2\text{p}$  metallic feature of the NiAl(100) substrate was centered at 72.6 eV.  $\text{NO}_2$  and  $\text{H}_2\text{O}$  used in the experiments were purified by several freeze–pump–thaw cycles.

### 3. Results and Discussion

**3.1. TPD Studies on  $\text{NO}_2/\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$  and  $\text{H}_2\text{O}/\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$ .** Figure 1a and b present TPD results for  $\text{NO}_2/\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$  and  $\text{H}_2\text{O}/\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$  adsorption systems for varying adsorbate surface coverages, respectively. A detailed interpretation of these results and the complimentary XPS data are given elsewhere.<sup>27,28</sup> For the  $\text{NO}_2/\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$  adsorption system (Figure 1a), XPS,<sup>28</sup> and TPD experi-

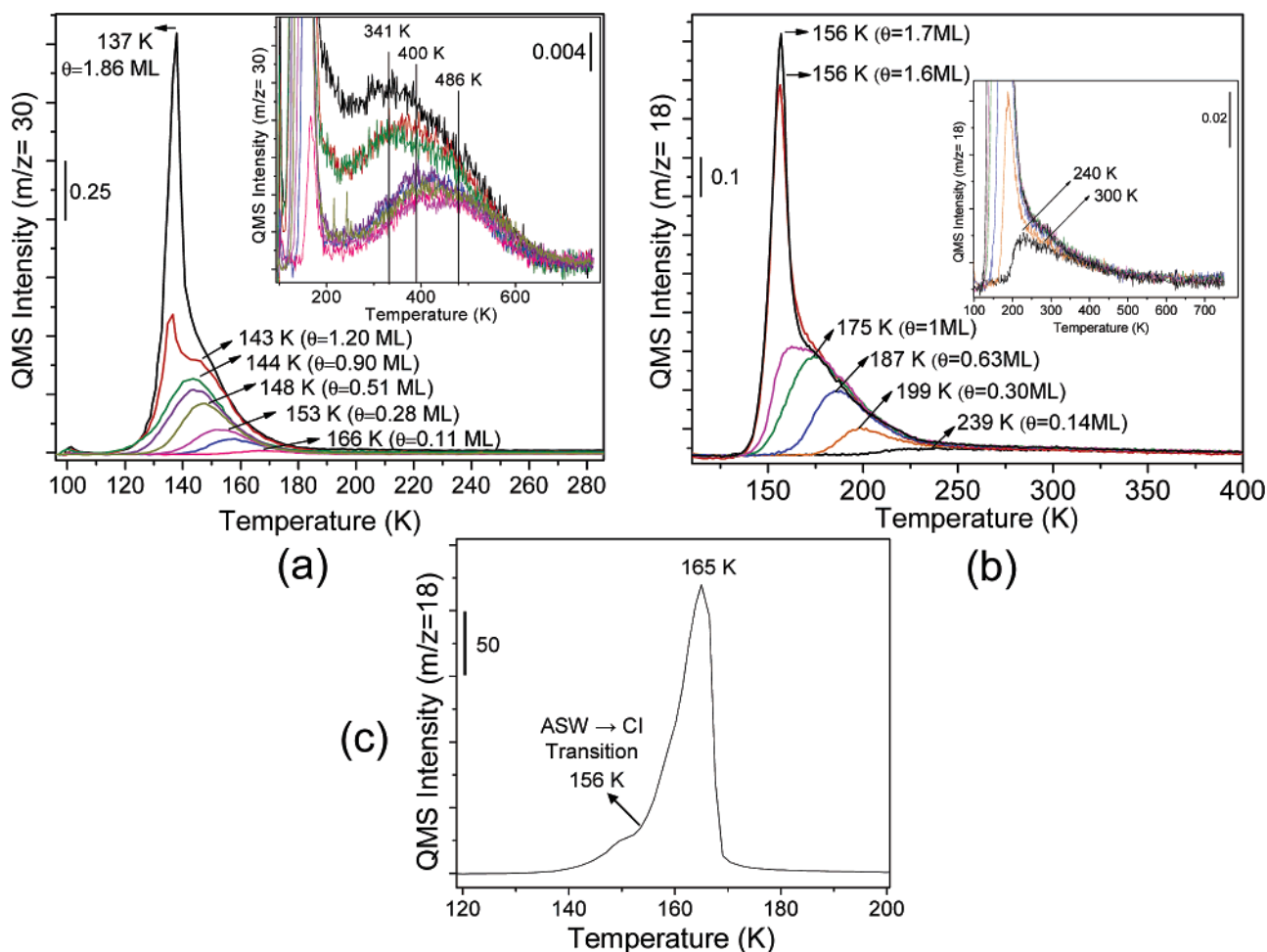
ments reveal that  $\text{NO}_2$ <sup>39</sup> adsorbs predominantly in a molecular fashion at low temperatures by first occupying surface cationic sites in the monolayer, resulting in a perturbed first-order desorption behavior with a temperature desorption maximum at  $144 \text{ K}$ <sup>40</sup> for  $\theta_{\text{NO}_2} = 1 \text{ ML}$  (ML = monolayer). After the completion of the monolayer, formation of physisorbed  $\text{NO}_2$  multilayers is evident which leads to desorption maxima at  $137 \text{ K}$  with a typical zero-order desorption characteristics, as frequently observed for weakly bound multilayer adsorption states. The results of previous XPS studies<sup>28</sup> revealed that physisorbed  $\text{NO}_2$  multilayers on  $\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$  existed in the form of  $\text{NO}_2$ -dimers (i.e.,  $\text{N}_2\text{O}_4$ ). Vibrational spectroscopic evidence for this assignment will be provided in the later section of the current text. Besides these molecular  $\text{NO}_2$  states, a smaller quantity ( $< 0.4 \text{ ML}$ ) of strongly bound  $\text{NO}_x$  surface species yielding TPD features within  $180\text{--}650 \text{ K}$  is visible as shown in the inset of Figure 1a. Based on the results of our XPS studies,<sup>28</sup> these features were assigned to ionic  $\text{NO}_x$  species, namely surface nitrites ( $\text{NO}_2^-$ ) and nitrates ( $\text{NO}_3^-$ ).

TPD data corresponding to  $\text{H}_2\text{O}$  adsorption on the  $\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$  surface, illustrated in Figure 1b, demonstrate that, similar to  $\text{NO}_2$  adsorption,  $\text{H}_2\text{O}$  adsorbs primarily in a molecular fashion to form, initially, a monolayer ( $175 \text{ K}$ ) and, subsequently, multilayers ( $156 \text{ K}$ ).<sup>27</sup> Desorption kinetics of  $\text{H}_2\text{O}$  on the  $\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$  surface shows, also very similarly to  $\text{NO}_2$ , perturbed first-order kinetics until the completion of the first monolayer, whereas the multilayer  $\text{H}_2\text{O}$  desorption takes place via zero-order kinetics as commonly observed on various solid surfaces.<sup>41,42</sup>  $\text{H}_2\text{O}$  desorption features appearing at  $T \geq 250 \text{ K}$  on oxide surfaces are commonly attributed to recombinative  $\text{H}_2\text{O}$  desorption, originating from hydroxyl groups that are formed due to the dissociation of  $\text{H}_2\text{O}$  molecules on oxide surfaces.<sup>42</sup> Therefore, the high temperature ( $250 \text{ K} < T < 600 \text{ K}$ ) region of the TPD spectra given in the inset of Figure 1b implies that only about  $0.10 \text{ ML}$  of  $\text{H}_2\text{O}$  molecules dissociatively adsorb on  $\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$ . Results of our vibrational spectroscopic studies also support the lack of extensive  $\text{H}_2\text{O}$  dissociation or formation of isolated hydroxyl groups on the  $\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$  surface.

The energetics of  $\text{H}_2\text{O}$  and  $\text{NO}_2$  adsorption can be discussed in a comparative manner in the light of the data given in Figure 1a and b. It is readily seen that considering the nondissociative (or molecular) adsorption states,  $\text{H}_2\text{O}$  adsorption on  $\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$  is stronger for both monolayer and multilayer states than that of  $\text{NO}_2$ . Since both  $\text{H}_2\text{O}$  and  $\text{NO}_2$  molecules follow first-order desorption kinetics for  $\theta_{\text{H}_2\text{O}}, \text{NO}_2 \leq 1 \text{ ML}$ , the Redhead approximation:<sup>43</sup>

$$\Delta E_{\text{des}} = RT_{\text{max}} \left[ \ln \frac{\nu T_{\text{max}}}{\beta} - 3.64 \right] \quad (1)$$

where  $\Delta E_{\text{des}}$  is the desorption energy,  $R$  is the universal gas constant,  $\nu$  is the preexponential coefficient,  $T_{\text{max}}$  is the desorption maxima for  $\theta_{\text{H}_2\text{O}}, \text{NO}_2 = 1 \text{ ML}$ ,  $\beta$  is the heating rate ( $\beta = 2 \text{ K}\cdot\text{s}^{-1}$ ), can be employed to estimate  $\Delta E_{\text{des}}$  values. If the values for the pre-exponential coefficients ( $\nu$ ) are assumed to be on the order of  $10^{13} \text{ s}^{-1}$  for both  $\text{H}_2\text{O}$  and  $\text{NO}_2$ ,  $\Delta E_{\text{des}}(\text{H}_2\text{O})_{\theta=1 \text{ ML}}$  and  $\Delta E_{\text{des}}(\text{NO}_2)_{\theta=1 \text{ ML}}$  can be calculated as  $44.8 \text{ kJ/mol}$  and  $36.6 \text{ kJ/mol}$ , respectively. The desorption energy values clearly show the difference in binding strength between  $\text{H}_2\text{O}$  and  $\text{NO}_2$ . It indicates that water adsorbs stronger onto the  $\theta\text{-Al}_2\text{O}_3$  surface than  $\text{NO}_2$ . Implications of this difference between the adsorption strengths will be further elaborated in Section 3.3 during the discussion of the  $(\text{NO}_2 + \text{H}_2\text{O})/\theta\text{-Al}_2\text{O}_3/\text{NiAl}(100)$  coadsorption system, where  $\text{NO}_2$  and  $\text{H}_2\text{O}$  competes



**Figure 1.** (a) 30 amu signal of the TPD data for various coverages of NO<sub>2</sub> on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100). Inset gives a detailed view of the high-temperature region of the 30 amu signal. (b) 18 amu signal of the TPD data for different H<sub>2</sub>O coverages on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100). Inset gives a detailed view of the high-temperature region of the 18 amu signal. (c) TPD data for 12 ML of H<sub>2</sub>O on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100). For each spectrum, adsorbates are introduced on a clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface at 80 K.

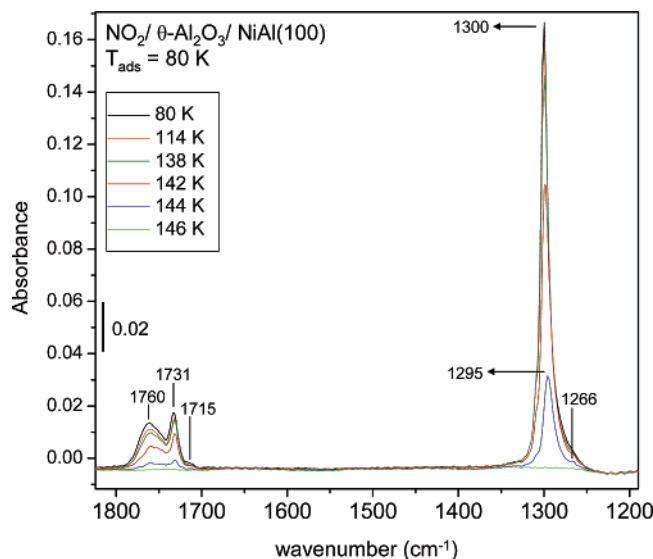
for the same adsorption sites on the oxide surface. It will be shown that despite the seemingly small difference in desorption energies ( $\Delta\Delta E_{\text{des}} = 8$  kJ/mol), H<sub>2</sub>O dominates by occupying the majority of the adsorption sites of the alumina surface leading to segregated three-dimensional NO<sub>2</sub> domains.

TPD data presented in Figure 1c highlights an important aspect of the thermally induced changes in the crystal structure of the H<sub>2</sub>O multilayers on solid surfaces. The TPD spectrum given in Figure 1c corresponds to a H<sub>2</sub>O coverage of  $\sim 12$  ML on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface. A readily visible break point and asymmetry in the low-temperature side of the multilayer H<sub>2</sub>O desorption feature is evident. This “pre-edge”, appearing around 156 K, has also been observed during the desorption of thick ( $\geq 5$  ML) H<sub>2</sub>O multilayers from a variety of solid surfaces<sup>41,42,44</sup> and has been associated with a phase transition from amorphous solid water (ASW) to crystalline ice (CI). Since ASW  $\rightarrow$  CI phase transition is an endothermic process, the water desorption rate in this temperature range decreases which is visible as a break point in the TPD profile. In other words, the difference between the desorption rates of ASW and CI results in the observation of a break in the TPD spectra at well-defined temperatures for given adsorbate coverages. Previous studies<sup>7,42,45,46</sup> suggested that for certain adsorption systems, the reactivity and the interaction of H<sub>2</sub>O layers with other adsorbed molecules may differ depending on the crystallinity of the solid H<sub>2</sub>O phase (i.e., ASW or CI). This point will be further

elaborated in Section 3.3 during the discussion of the TPD data for the NO<sub>2</sub> + H<sub>2</sub>O coadsorption system on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100).

**3.2. IRAS Studies on NO<sub>2</sub>/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) and NO<sub>2</sub>/H<sub>2</sub>O/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100).** Figure 2 illustrates IRAS results obtained for the low-temperature (80–146 K) NO<sub>2</sub> adsorption on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100). The series of spectra given in Figure 2 is obtained by dosing multilayers of NO<sub>2</sub> on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface at 80 K and, subsequently, annealing the sample to the given temperatures for a few seconds. After each annealing step, the sample was cooled back to 80 K for data acquisition. The major vibrational features observed upon NO<sub>2</sub> deposition appear as a very strong band at 1300 cm<sup>-1</sup> and two additional strong bands at 1731 and 1760 cm<sup>-1</sup>. Besides these prominent features, a low frequency shoulder at 1266 cm<sup>-1</sup> and a less intense feature at 1715 cm<sup>-1</sup> can be seen in Figure 2. Based on the results of previous NO<sub>2</sub> adsorption studies,<sup>5–8,18,19,22,24,26</sup> we assign the 1266 and 1300 cm<sup>-1</sup> bands to the  $\nu_s(\text{NO}_2)$  or NO<sub>2</sub> symmetric stretch of N<sub>2</sub>O<sub>4</sub> monolayer and multilayers, respectively. Likewise, the bands at 1715, 1731, and 1760 cm<sup>-1</sup> are assigned to the NO<sub>2</sub> asymmetric stretch,  $\nu_a(\text{NO}_2)$ , of N<sub>2</sub>O<sub>4</sub> monolayer and multilayers. It should be noted that the weak feature at 1266 cm<sup>-1</sup> could also be attributed to the presence of N<sub>2</sub>O<sub>3</sub>(a) species which may form in the reaction of background NO with the NO<sub>2</sub> gas. N<sub>2</sub>O<sub>3</sub> adsorbed on solid surfaces results in IRAS active vibrational bands at 1265 cm<sup>-1</sup> ( $\nu_a(\text{NO}_2)$ ) and 1900 cm<sup>-1</sup> ( $\nu(\text{NO})$ ).<sup>8,22</sup> However, no bands at  $\sim 1900$  cm<sup>-1</sup>





**Figure 2.** IRAS spectra for NO<sub>2</sub> multilayers on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface at different temperatures. Initial NO<sub>2</sub> exposure was made at 80 K. After the adsorbate exposure, the sample surface was annealed to the given temperatures and cooled back to 80 K for data acquisition.

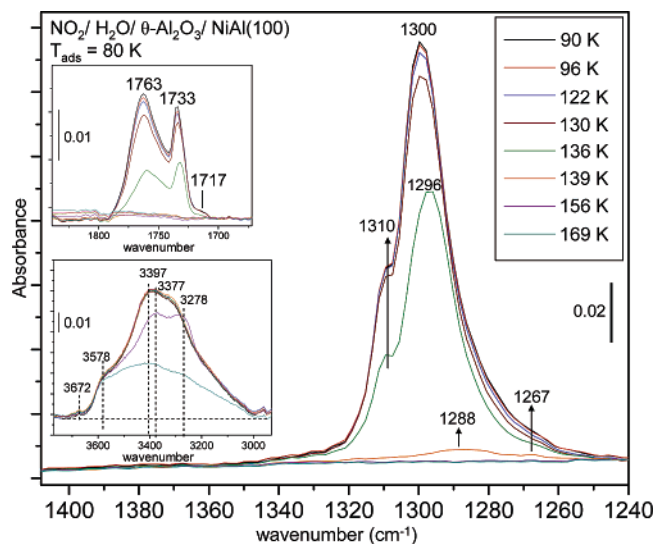
were visible in our experiments, suggesting that the surface coverage of N<sub>2</sub>O<sub>3</sub> is below our detection limit.

Further information regarding the orientation of the N<sub>2</sub>O<sub>4</sub> multilayers on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface can be deduced by comparing the relative intensities of  $\nu_s$ (NO<sub>2</sub>) (1266–1300 cm<sup>-1</sup>) and  $\nu_a$ (NO<sub>2</sub>) (1715–1760 cm<sup>-1</sup>) bands, taking advantage of the fact that, according to the surface selection rules for infrared radiation reflecting from a conducting solid, only the vibrational modes having a dynamic dipole moment oriented perpendicular to the surface are excited by the incident IR radiation.<sup>47</sup> The observed intensity ratios for the symmetric to asymmetric NO<sub>2</sub> stretches, ( $I_{\nu_s}$ )/( $I_{\nu_a}$ ), thus reveal that N<sub>2</sub>O<sub>4</sub> possesses *D*<sub>2h</sub> symmetry<sup>5–7</sup> where the N–N bond is perpendicular to the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface. This is based on the fact that the dynamic dipole moment of the  $\nu_s$ (NO<sub>2</sub>) mode of *D*<sub>2h</sub>-N<sub>2</sub>O<sub>4</sub> is directed parallel to the N–N bond while that of  $\nu_a$ (NO<sub>2</sub>) is oriented perpendicular to the N–N bond.

It should also be mentioned that the current vibrational spectroscopic results indicating the dimerization of NO<sub>2</sub> to form multilayer N<sub>2</sub>O<sub>4</sub> layers on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface at low temperatures are consistent with the results of our previous XPS experiments,<sup>28</sup> suggesting the presence of a nonionic N<sub>2</sub>O<sub>4</sub> isomer structure. It is known from the literature<sup>5,48–51</sup> that N<sub>2</sub>O<sub>4</sub> can also exist in the form of D-isomers, O=N–O–NO<sub>2</sub>, which can transform into NO<sup>+</sup> NO<sub>3</sub><sup>-</sup> (nitrosonium nitrate) at higher temperatures. The characteristic vibrational bands for D-isomers are  $\nu_s$ (NO<sub>2</sub>) at ~1300 cm<sup>-1</sup>,  $\nu_a$ (NO<sub>2</sub>) at ~1640 cm<sup>-1</sup>, and  $\nu$ -(NO) at ~1820 cm<sup>-1</sup>. Since the bands at 1640 and 1820 cm<sup>-1</sup> are not observed in our IRAS results, existence of D-isomers or nitrosonium nitrate can be excluded.

In line with the TPD data presented in Figure 1a, Figure 2 shows that NO<sub>2</sub> multilayers in the form of N<sub>2</sub>O<sub>4</sub> are weakly adsorbed on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface. IRAS signal originating from the multilayers decreases with increasing surface temperatures and almost completely disappears at ca. 146 K, during which the strong  $\nu_s$ (NO<sub>2</sub>) band at 1300 cm<sup>-1</sup> progressively redshifts to lower frequencies.<sup>52</sup>

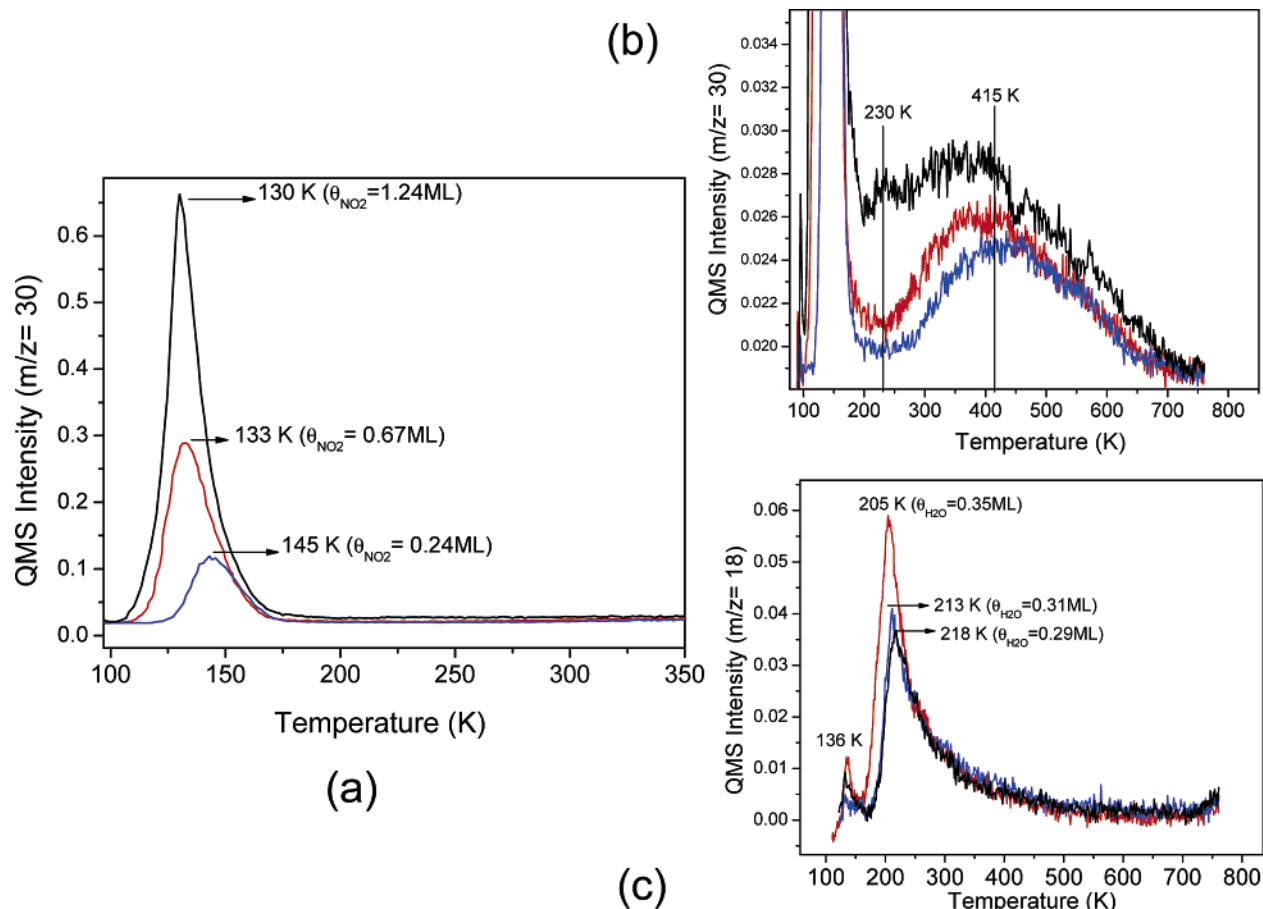
Figure 3 illustrates IRAS results for NO<sub>2</sub> multilayers deposited on a thick (>10 ML) H<sub>2</sub>O layer dosed on a clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) substrate. Sections of the vibrational spectra relevant to  $\nu_a$ (NO<sub>2</sub>) modes (1650–1850 cm<sup>-1</sup>) of NO<sub>2</sub>/N<sub>2</sub>O<sub>4</sub> and  $\nu$ (OH)



**Figure 3.**  $\nu_s$ (NO<sub>2</sub>) region of the IRAS spectra for NO<sub>2</sub> multilayers deposited on a  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface which is initially predosed with multilayers of H<sub>2</sub>O. Insets provide detailed views of the  $\nu_a$ (NO<sub>2</sub>) and  $\nu$ (OH) regions. After the adsorbate exposures, the sample surface was annealed to the given temperatures and cooled back to 80 K for data acquisition.

modes (3000–4000 cm<sup>-1</sup>) of H<sub>2</sub>O are given as insets. Similar to the NO<sub>2</sub> multilayers on the clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface, vibrational bands at 1267, 1300 cm<sup>-1</sup> associated with  $\nu_s$ (NO<sub>2</sub>) modes of N<sub>2</sub>O<sub>4</sub> and bands at 1717, 1733, and 1763 cm<sup>-1</sup> associated with the  $\nu_a$ (NO<sub>2</sub>) modes of N<sub>2</sub>O<sub>4</sub> multilayers are also visible for NO<sub>2</sub>/H<sub>2</sub>O/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) system. In addition to these features, a readily visible shoulder at 1310 cm<sup>-1</sup> is also observed in the  $\nu_s$ (NO<sub>2</sub>) region. As this feature emerges in the presence of H<sub>2</sub>O layers, it is most likely linked to the N<sub>2</sub>O<sub>4</sub> layers that are interacting with the underlying H<sub>2</sub>O layer resulting in a slightly different N<sub>2</sub>O<sub>4</sub> orientation with respect to the isolated N<sub>2</sub>O<sub>4</sub> layers. Support for a different orientation for the N<sub>2</sub>O<sub>4</sub> layers interacting with the underlying H<sub>2</sub>O layers comes from the differences in the relative intensities of the  $\nu_a$ -(NO<sub>2</sub>) features at 1730 and 1760 cm<sup>-1</sup> in the presence and absence of H<sub>2</sub>O. It is apparent that the presence of H<sub>2</sub>O leads to an increase in the intensity of the 1763 cm<sup>-1</sup> band with respect to the intensity of 1733 cm<sup>-1</sup> feature. IRAS data in Figure 3 also imply that, although some part of the N<sub>2</sub>O<sub>4</sub> layer is interacting and, therefore, being affected by the presence of an underlying H<sub>2</sub>O layer, isolated N<sub>2</sub>O<sub>4</sub> islands still exist, whose  $\nu_s$ (NO<sub>2</sub>) bands are located at 1300 cm<sup>-1</sup>. These observations are also consistent with the changes in the fwhm of the  $\nu_s$ (NO<sub>2</sub>) band in the presence and absence of H<sub>2</sub>O (11 cm<sup>-1</sup> in Figure 2 vs 17 cm<sup>-1</sup> in Figure 3), suggesting that the presence of H<sub>2</sub>O layers introduces additional adsorption sites/orientations for the N<sub>2</sub>O<sub>4</sub> species resulting in a broadening of their  $\nu_s$ (NO<sub>2</sub>) bands. The spectrum corresponding to 139 K in Figure 3 suggests that as the N<sub>2</sub>O<sub>4</sub> multilayer thickness decreases,  $\nu_s$ (NO<sub>2</sub>) band which was initially observed at 1300 cm<sup>-1</sup> for thicker N<sub>2</sub>O<sub>4</sub> layers shifts to 1288 cm<sup>-1</sup> and approaches the position of the weak  $\nu_s$ (NO<sub>2</sub>) mode at 1266 cm<sup>-1</sup>, indicating that the 1266 cm<sup>-1</sup> feature can be attributed to NO<sub>2</sub>/N<sub>2</sub>O<sub>4</sub> molecules with a coverage close to 1 ML.

The lower inset of Figure 3 presents the section of the IRAS data where  $\nu$ (OH) modes are detected. A broad and highly convoluted  $\nu$ (OH) stretching band appears after the deposition of a thick H<sub>2</sub>O layer at 80 K lies within 3300–3400 cm<sup>-1</sup> and corresponds to fully coordinated H<sub>2</sub>O molecules via hydrogen bonding in the bulk ASW network.<sup>42</sup> Influence of the ASW →



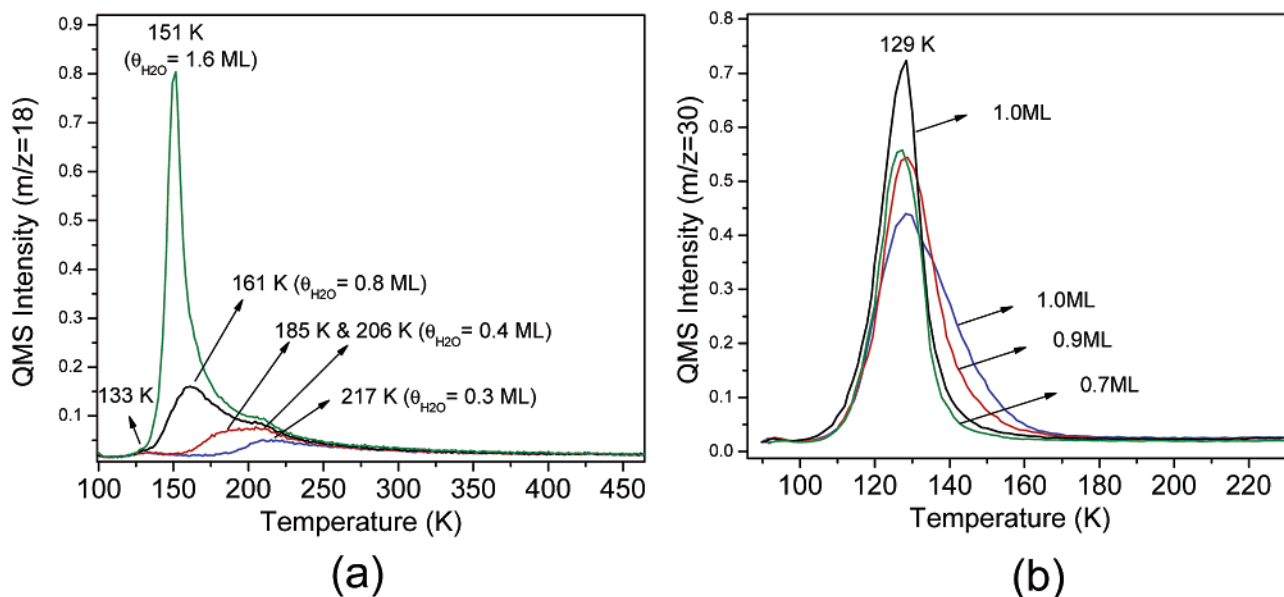
**Figure 4.** TPD data for NO<sub>2</sub> adsorption on a  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface which is initially exposed to submonolayer (0.3 ML) coverages of H<sub>2</sub>O. (a) 30 amu signal, (b) detailed view of the high-temperature tail of the 30 amu signal, (c) 18 amu signal. All of the adsorbate exposures were performed at a substrate temperature of 80 K.

CI phase transition on the  $\nu(\text{OH})$  modes of water layers can also be monitored in the IRA spectrum obtained at  $T = 156$  K, revealing a significant ordering in the  $\nu(\text{OH})$  bands at 3300–3400  $\text{cm}^{-1}$  and a red shift of  $\sim 20$   $\text{cm}^{-1}$ . An important aspect of the vibrational spectroscopic data given in the lower inset is the minor feature visible at 3672  $\text{cm}^{-1}$ . In the literature<sup>42,44</sup> such features have been attributed to  $\nu(\text{OH})$  modes originating from uncoordinated OH bonds on the upper surface of a H<sub>2</sub>O layer, H<sub>2</sub>O molecules at the periphery or interior of two-dimensional water structures on solid substrates, or dangling OH bonds on the surface of 2D or 3D H<sub>2</sub>O clusters. Data presented in the current work does not allow us to deduce a conclusive model for the exact surface morphology of the H<sub>2</sub>O layer that is in contact with the NO<sub>2</sub> multilayers at 80 K on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) substrate. Further detailed scanning probe experiments or low-temperature ( $< 80$  K) physisorption studies using weakly binding probe molecules such as N<sub>2</sub> or Kr would be helpful to elucidate this aspect. However, it is worth mentioning that we have not observed a significant increase in the intensity of the feature at 3672  $\text{cm}^{-1}$  upon annealing the surface to  $T > 80$  K, implying that the majority of the adsorbed H<sub>2</sub>O molecules adsorb and desorb in a molecular fashion consistent with our previous TPD results.<sup>27</sup> From this respect, our results differ from those which suggest dissociative adsorption of H<sub>2</sub>O on Al<sub>2</sub>O<sub>3</sub>/Al(111)<sup>53</sup> and on Mo(110)–(1  $\times$  6)–O<sup>26</sup> surfaces at  $T > 150$  K leading to intense and well-resolved  $\nu(\text{OH})$  features at 3720 and 3584  $\text{cm}^{-1}$ , respectively.

**3.3. TPD Studies on NO<sub>2</sub>/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) and NO<sub>2</sub>/H<sub>2</sub>O/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100).** The interaction between NO<sub>2</sub> and H<sub>2</sub>O on a clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) substrate is studied via the TPD

technique by sequential deposition of these probe molecules on the sample surface using various preparation protocols. In these TPD experiments, relative coverages of each adsorbate, deposition order, and deposition temperatures were varied. A selection of the TPD results that are representative of the overall characteristics of the NO<sub>2</sub> + H<sub>2</sub>O adsorption system on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface is discussed below. During the TPD experiments, various possible NO<sub>2</sub> + H<sub>2</sub>O reaction/desorption products such as 18 amu (H<sub>2</sub>O), 28 amu (N<sub>2</sub>), 30 amu (NO, NO<sub>2</sub>, H<sub>x</sub>N<sub>y</sub>O<sub>z</sub>), 31 amu (HNO, H<sub>x</sub>N<sub>y</sub>O<sub>z</sub>), 32 amu (O<sub>2</sub>), 44 amu (N<sub>2</sub>O), 45 amu (HN<sub>2</sub>O), 46 amu (NO<sub>2</sub>), 47 amu (HONO), and 63 amu (HNO<sub>3</sub>) were monitored. In the TPD experiments, the major desorption products were H<sub>2</sub>O, NO<sub>2</sub>, and NO. Besides these major products, smaller H<sub>x</sub>N<sub>y</sub>O<sub>z</sub> desorption signals were also detected similar to those that were reported on NO<sub>2</sub>/H<sub>2</sub>O/Au(111).<sup>7</sup> Intensities of these minor signals were 2 orders of magnitude or much smaller than the major desorption products which were simultaneously detected. Control experiments revealed that the quantities of these minority products were not reproducible. Therefore, they are attributed to background reactions in the QMS during the TPD experiments.

The influence of the presence of submonolayer amounts of H<sub>2</sub>O on NO<sub>2</sub> adsorption was examined in the series of TPD experiments shown in Figure 4. In these experiments various amounts of NO<sub>2</sub> (0.2 ML  $< \theta_{\text{NO}_2} < 1.2$  ML) were adsorbed on a H<sub>2</sub>O/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface which was initially predosed with 0.3 ML of H<sub>2</sub>O. In this deposition method, both of the adsorbates were introduced onto the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface at 80 K. As it was described in section 3.1,<sup>28,39</sup> the major NO<sub>x</sub> desorption signal observed was 30 amu, which originated from



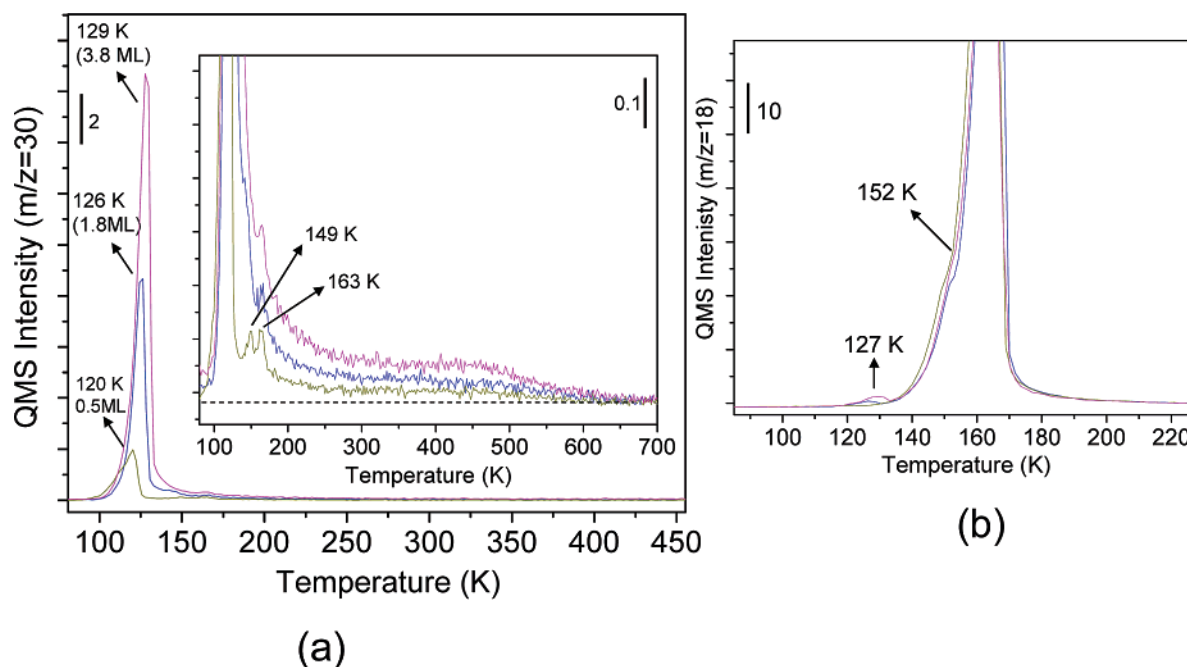
**Figure 5.** TPD data for various H<sub>2</sub>O coverages (0.3–1.6 ML) on a  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface which is initially exposed to  $\sim 1$  ML of NO<sub>2</sub>. (a) 18 amu signal. (b) 30 amu signal. All of the adsorbate exposures were performed at a surface temperature of 80 K.

NO<sub>2</sub> (Figure 4a) and other ionic NO<sub>x</sub> species (Figure 4b).<sup>28</sup> The 18 amu signal resulting from H<sub>2</sub>O desorption is presented in Figure 4c. Comparison of molecular NO<sub>2</sub> desorption features given in Figures 4a and 1a reveals that the desorption maxima in Figure 4a (corresponding to molecular NO<sub>2</sub> adsorption on a H<sub>2</sub>O pre-dosed surface) shifts to slightly lower temperatures ( $\sim 7$  K) for the given NO<sub>2</sub> coverages with respect to that of Figure 1a (NO<sub>2</sub> adsorption on the clean substrate). Furthermore, by comparing the TPD traces corresponding to a NO<sub>2</sub> coverage of 1.2 ML in Figures 1a and 4a, it is evident that in the absence of H<sub>2</sub>O, the multilayer NO<sub>2</sub> feature develops only after the saturation of the first layer, therefore, the monolayer and the multilayer features are easily distinguished in the TPD spectrum. On the other hand, in the presence of  $\sim 0.3$  ML of H<sub>2</sub>O, such a distinct separation between monolayer and multilayer NO<sub>2</sub> features is not apparent. This observation suggests that the presence of submonolayer (0.3 ML) quantities of H<sub>2</sub>O changes the growth mode of NO<sub>2</sub> layers on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface. On a clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) substrate, increasing NO<sub>2</sub> coverage resembles a layer by layer type of growth, whereas in the presence of submonolayer quantities of H<sub>2</sub>O, NO<sub>2</sub> layers grow in a form that resembles 3D clusters. These results indicate a competition for the adsorption sites between the NO<sub>2</sub> and H<sub>2</sub>O molecules on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100). Since H<sub>2</sub>O binds stronger to the alumina surface than NO<sub>2</sub>, H<sub>2</sub>O molecules occupy the energetically more favorable adsorption sites and thus make them unavailable for NO<sub>2</sub> adsorption. At low NO<sub>2</sub> coverages, NO<sub>2</sub> binds to the adsorption sites on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface that are free of H<sub>2</sub>O. However, since these sites are energetically not as favorable as those present in the absence of H<sub>2</sub>O, the NO<sub>2</sub> desorption temperature shifts to lower values. At higher NO<sub>2</sub> coverages, the NO<sub>2</sub> desorption feature in Figure 4a is a composite of several origins: NO<sub>2</sub> directly bound to the available sites on the alumina surface, on top of weakly bound first layer NO<sub>2</sub>, and NO<sub>2</sub> adsorbed on top of H<sub>2</sub>O. The result of these multiple desorption processes is a broad, poorly resolved desorption feature. The lower temperature of the molecular NO<sub>2</sub> desorption maxima observed in the presence of H<sub>2</sub>O with respect to that of the H<sub>2</sub>O-free surface suggests the destabilization of the NO<sub>2</sub> multilayer by the presence of H<sub>2</sub>O.

Figure 4b shows the formation of strongly bound NO<sub>x</sub> species with a coverage of  $\theta_{\text{NO}_x} < 0.25$  ML on the H<sub>2</sub>O(0.3 ML)/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface. According to the discussion above, although H<sub>2</sub>O blocks some of the surface sites, a large number of adsorption sites remain available for NO<sub>2</sub> adsorption where the formation of strongly bound surface nitrates and nitrites occurs. This argument is also supported by the current XPS results to be presented in section 3.4. Water desorption signals can also be seen in more detail in Figure 4c. Comparison of Figures 1b and 4c reveals that for  $\theta_{\text{H}_2\text{O}} = 0.3$  ML, H<sub>2</sub>O desorption features are shifted to higher ( $+15$  K) temperatures in the presence of NO<sub>x</sub> species. Since at  $T > 200$  K, NO<sub>x</sub> species existing on the surface are in the form of nitrates and nitrites,<sup>28</sup> the stabilization of the adsorbed H<sub>2</sub>O molecules can be attributed to its interaction with ionic NO<sub>x</sub> species on the surface. An additional small H<sub>2</sub>O desorption feature at 136 K is observed in Figure 4c which is not present in Figure 1b. Figure 4a shows that this temperature is very close to the NO<sub>2</sub> desorption features from the H<sub>2</sub>O(0.3 ML)/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface. Therefore, this feature is tentatively assigned to a small quantity of H<sub>2</sub>O molecules that are trapped in the NO<sub>2</sub> matrix. It should be emphasized that despite the presence of this small portion of H<sub>2</sub>O molecules that are trapped in the NO<sub>2</sub> matrix, H<sub>2</sub>O and molecular NO<sub>2</sub> domains seem to be mostly segregated for  $\theta_{\text{H}_2\text{O}} \leq 0.3$  ML and  $\theta_{\text{NO}_2} \leq 1.2$  ML where H<sub>2</sub>O adsorbs more strongly on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface and forces NO<sub>2</sub> to form 2D and 3D clusters.

The coadsorption of H<sub>2</sub>O and NO<sub>2</sub> on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub> thin film was also investigated by first exposing the clean surface to NO<sub>2</sub> (where the NO<sub>2</sub> coverage is kept almost constant at  $\sim 1$  ML), and then to varying amounts of H<sub>2</sub>O (0.3 ML–1.6 ML), both at 80 K. 18 and 30 amu channels of these TPD experiments are given in Figure 5. Figure 5a shows that monolayer to multilayer transitions are observed for the H<sub>2</sub>O layers as in Figure 1b; however, compared with the H<sub>2</sub>O adsorption states on a clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface, additional H<sub>2</sub>O adsorption states appear, particularly for  $\theta_{\text{H}_2\text{O}} \geq 0.4$  ML. Examining the  $\theta_{\text{H}_2\text{O}} = 0.4$  ML signal given in Figure 5a reveals two poorly resolved but distinguishable desorption features at 185 and 206 K. The behavior of the lower temperature feature (185 K) resembles that observed for H<sub>2</sub>O desorption from the clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface





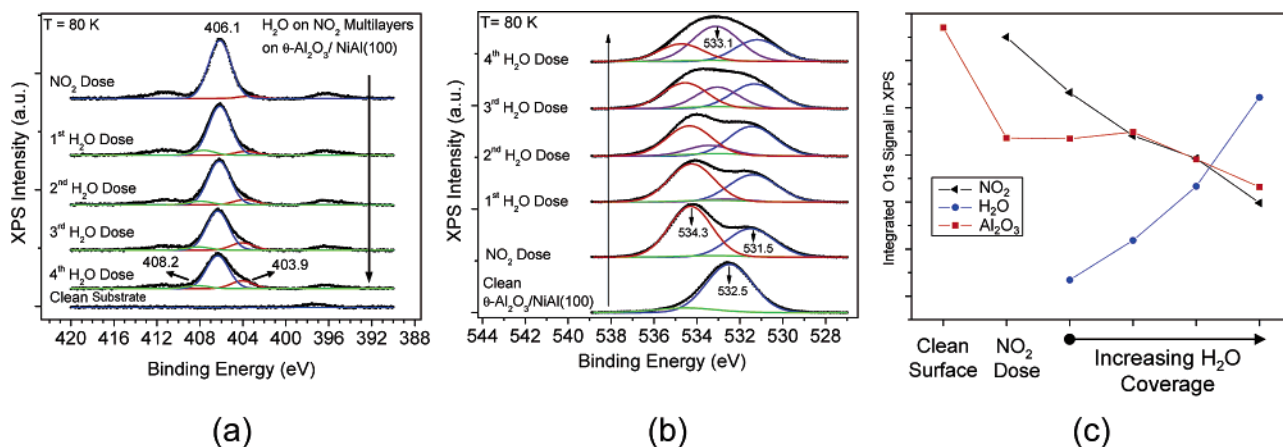
**Figure 6.** TPD data for NO<sub>2</sub> adsorption on a  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface which is initially exposed to 10 ML of H<sub>2</sub>O. (a) 30 amu signal where the inset depicts a detailed view of the high-temperature tail of the 30 amu signal. (b) 18 amu signal. All of the adsorbate exposures were performed at a surface temperature of 80 K.

at similar coverages (Figure 1b). The higher temperature desorption feature (206 K), however, appears at a temperature which is  $\sim 7$  K higher than the one observed on the clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface. Interestingly, this high-temperature feature saturates at water coverages between 0.3 and 0.4 ML, coinciding with the coverages of ionic NO<sub>x</sub> species (NO<sub>2</sub><sup>-</sup> and NO<sub>3</sub><sup>-</sup>) on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface.<sup>28</sup> This observation suggests that new, energetically favorable H<sub>2</sub>O adsorption sites are created on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface after its exposure to NO<sub>2</sub>. Therefore, these two low-temperature H<sub>2</sub>O desorption features at 206 and 185 K can be assigned to H<sub>2</sub>O adsorbed on alumina sites associated with the presence of ionic NO<sub>x</sub> species, and H<sub>2</sub>O adsorbed onto NO<sub>x</sub>-free alumina sites, respectively. Due to its higher strength of adsorption relative to molecular NO<sub>2</sub> adsorption, H<sub>2</sub>O may replace most of the adsorbed NO<sub>2</sub> molecules on the alumina surface, and directly interact with the alumina film. After the completion of the first H<sub>2</sub>O layer on various adsorption sites, H<sub>2</sub>O multilayers are formed, as evident from the desorption feature at 151 K.

There are noticeable changes in the desorption peak shape of the molecularly adsorbed NO<sub>2</sub> with increasing H<sub>2</sub>O coverage, as shown in Figure 5b. At low water coverages ( $\theta_{\text{H}_2\text{O}} = 0.3$ –0.4 ML), NO<sub>2</sub> and H<sub>2</sub>O islands directly bound to the  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface coexist; therefore, a high temperature shoulder (130–160 K) in the NO<sub>2</sub> desorption feature is present. Notably, the desorption temperature of this high-temperature NO<sub>2</sub> shoulder is very similar to that seen in Figure 4a when a low coverage of NO<sub>2</sub> was adsorbed onto  $\sim 0.3$  ML of H<sub>2</sub>O. As the water coverage increases, this higher temperature NO<sub>2</sub> desorption feature disappears, and at a  $\sim 1$  ML H<sub>2</sub>O dose, there is only one sharp NO<sub>2</sub> desorption peak at 129 K, which is located at a lower temperature than that observed for NO<sub>2</sub> multilayer desorption from a clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface. Note that this low-temperature feature (129 K) is the dominant NO<sub>2</sub> peak even at  $\theta_{\text{H}_2\text{O}} < 1$  ML. Therefore, the 129 K feature is attributed to molecularly adsorbed 3D NO<sub>2</sub> clusters weakly bound to the H<sub>2</sub>O adlayer modified  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface.

Next, we investigated the adsorption of NO<sub>2</sub> on a  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface that is fully covered by H<sub>2</sub>O molecules. For this purpose, TPD experiments were carried out by depositing NO<sub>2</sub> with varying coverages on a  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface that had been dosed with  $\sim 10$  ML of H<sub>2</sub>O at 80 K prior to NO<sub>2</sub> exposure. The results of these experiments are illustrated in Figure 6. Figure 6a depicts the 30 amu signal originating from NO<sub>x</sub> desorption. A detailed view of the high-temperature section of the 30 amu signal is given in the inset. Zero-order desorption behavior of NO<sub>2</sub> multilayers leading to temperature maxima at 120, 126, and 129 K with increasing  $\theta_{\text{NO}_2}$  is apparent in Figure 6a. On these water-covered  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surfaces, regardless of the NO<sub>2</sub> surface coverage, NO<sub>2</sub> forms 3D islands as the NO<sub>2</sub> desorption feature resembles the TPD trace seen for multilayer NO<sub>2</sub> on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>. This is due to the weak interaction between NO<sub>2</sub> and the 10 ML-thick H<sub>2</sub>O film.

An important aspect of the NO<sub>2</sub> desorption from the H<sub>2</sub>O(10 ML)/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface can be seen in the inset of Figure 6a. Two sharp and distinctly resolved NO<sub>2</sub> desorption features are observed at 149 and 163 K. Note that the intensities of these two new desorption features are independent of the NO<sub>2</sub> coverage. Also, these desorption features were completely absent in TPD spectra when NO<sub>2</sub> desorption was studied at low H<sub>2</sub>O coverages ( $< 2$  ML), suggesting that they are closely related to the presence of a thick H<sub>2</sub>O multilayer on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface. Furthermore, these desorption phenomena, observed in the presence of thick H<sub>2</sub>O overlayers occur only in a very narrow temperature window which directly overlaps with the ASW  $\rightarrow$  CI phase transition. Therefore, we attribute these two distinct NO<sub>2</sub> desorption features to the evolution of NO<sub>2</sub> due to this phase transition of the H<sub>2</sub>O layer. A number of different factors can be considered to explain the evolution of NO<sub>2</sub> during the crystallization of the amorphous solid H<sub>2</sub>O layer. One of these arguments is the simple reduction in surface area or loss of porosity of the underlying H<sub>2</sub>O layer due to the ordering of the ASW to form CI. Another possible factor is the changes in the coordination of the water molecules and the hydrogen bonding at the surface and/or inside the ice matrix during the phase



**Figure 7.** XPS results for successive H<sub>2</sub>O adsorption on a  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface which is initially covered with multilayers of NO<sub>2</sub> ( $\sim 3$  ML). (a) N1s signal. (b) O1s signal. (c) intensities of the O1s signals originating from various surface species. All of the adsorbate exposures and data acquisition were performed at 80 K. Estimated H<sub>2</sub>O surface coverages after the given H<sub>2</sub>O deposition steps are 0.5ML, 1.0, 2.0, and 3.0 ML.

transformation which may ultimately modify the quantity of the NO<sub>2</sub> molecules that can be solvated/mixed or trapped in/on the H<sub>2</sub>O film. Some mixing of NO<sub>2</sub> and H<sub>2</sub>O layers in the presence of multilayer water is also suggested by the small H<sub>2</sub>O desorption feature at 127 K in Figure 6b. The ASW  $\rightarrow$  CI phase transition can clearly be seen near 152 K during the desorption of NO<sub>2</sub> and H<sub>2</sub>O from the NO<sub>2</sub>/H<sub>2</sub>O(10ML)/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) system. Changes in the extent of solvation or reactivity of H<sub>2</sub>O as a function of the water crystal structure have also been previously reported. For example, using TOF-SIMS and TPD techniques, Souda<sup>45</sup> recently reported that the solubility of octane in amorphous solid water is much higher than that of liquid water. Along these lines, NMR and X-ray diffraction studies of Tulk, et al.<sup>46</sup> showed that tetrahydrofuran (THF), a water soluble cyclic ether, is forced out of the amorphous solid water solution resulting in separated THF and H<sub>2</sub>O phases as a consequence of the ASW  $\rightarrow$  CI transition. Furthermore, Wang and Koel<sup>7</sup> showed that for NO<sub>2</sub>/H<sub>2</sub>O coadsorption on a Au(111) surface, O<sub>2</sub> desorption is only observed in the presence of ASW but not on CI.

Another important point regarding Figure 6a is the desorption of the strongly bound NO<sub>x</sub> species appearing within 200–650 K. As was discussed previously in our XPS study,<sup>28</sup> this broad desorption feature originates from ionic NO<sub>x</sub> species, namely nitrites and nitrates. It is important to emphasize that these species are formed even in the presence of a 10 ML thick H<sub>2</sub>O layer, which blocks a great majority (most likely all) of the available adsorption sites on the alumina surface where NO<sub>2</sub> can directly adsorb. Therefore, these strongly bound NO<sub>x</sub> species observed on the H<sub>2</sub>O(10 ML)/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface are either formed via solvation and diffusion of NO<sub>2</sub> in the ice layer and eventual adsorption on the alumina surface during the TPD experiment after the desorption of the thick H<sub>2</sub>O multilayer, or via disproportionation reactions of NO<sub>2</sub> leading to nitrite and nitrate formation on the ice surface and subsequent migration of these ionic NO<sub>x</sub> species to the alumina surface after the desorption of the H<sub>2</sub>O multilayer.

**3.4. XPS Studies on H<sub>2</sub>O/NO<sub>2</sub>/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100).** Figure 7 presents XPS results for a H<sub>2</sub>O + NO<sub>2</sub> coadsorption system on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) where various coverages of H<sub>2</sub>O ( $\theta_{\text{H}_2\text{O}} = 0.5, 1.0, 2.0, 3.5$  ML, denoted as steps 1–4 in the figure) are adsorbed onto NO<sub>2</sub> multilayers ( $\theta_{\text{NO}_2} \approx 3$  ML) at 80 K. Note that the coverage values for the XPS data are based on exposures and not additionally calibrated via TPD. N1s and O1s regions of the XP spectra are given in Figures 7a and 7b, respectively. Intensities of the O1s signals originating from different NO<sub>x</sub>,

H<sub>2</sub>O and Al<sub>2</sub>O<sub>3</sub> species are also plotted in Figure 7c for different H<sub>2</sub>O/NO<sub>2</sub> doses. The topmost spectrum in Figure 7a corresponds to NO<sub>2</sub> multilayers deposited on a clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) substrate. This spectrum is dominated by a N1s feature at 406.1 eV which is assigned to molecularly adsorbed NO<sub>2</sub>/N<sub>2</sub>O<sub>4</sub> species.<sup>28</sup> In addition to this, an additional minor feature is present at 403.9 eV which can be attributed to nitrite species.<sup>28</sup> As H<sub>2</sub>O is introduced onto the NO<sub>2</sub>/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface, the molecular NO<sub>x</sub> signal starts to attenuate. This is due to the formation of 3D NO<sub>2</sub> clusters forming on the water overlayer. As we have discussed previously, H<sub>2</sub>O, due to its stronger affinity for the  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface than NO<sub>2</sub>, can readily diffuse through the NO<sub>2</sub> multilayer. As the H<sub>2</sub>O/ $\theta$ -Al<sub>2</sub>O<sub>3</sub> interface develops, the NO<sub>2</sub> multilayer forms 3D islands on top of the H<sub>2</sub>O layer, resulting in a decrease of the N1s signal of these NO<sub>2</sub> clusters with increasing H<sub>2</sub>O coverage. H<sub>2</sub>O deposition also leads to the development of a new N1s feature at 408.2 eV which can be assigned to nitrates. Increasing the H<sub>2</sub>O exposure results in a slight increase in the intensity of the nitrite species, whereas the nitrate concentration remains steady (Figure 7a). Changes in the O1s region of the XP spectra can also provide additional support for the discussion above. As seen in Figure 7b, the O1s signal for the clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) substrate is dominated by a feature at 532.5 eV. Introduction of NO<sub>2</sub> multilayers induces a  $-1.0$  eV binding energy shift in the O1s features of the alumina substrate<sup>28</sup> and the O1s feature associated with the NO<sub>2</sub> layers appears at 534.3 eV. A large attenuation of the substrate O 1s feature upon the deposition of NO<sub>2</sub> multilayers is clearly evident in Figure 7c. Introduction of the first H<sub>2</sub>O dose results in an intensity drop of the molecularly adsorbed NO<sub>2</sub> feature (534.3 eV), while that of the substrate alumina film (531.5 eV) stays almost constant. This observation indicates that the NO<sub>2</sub> multilayers start to form 3D clusters as soon as the first H<sub>2</sub>O dose is introduced. This is even more evident in the XP spectrum collected after the introduction of the second H<sub>2</sub>O dose. The intensity of the O1s of molecularly adsorbed NO<sub>2</sub> decreases significantly, while the peak due to adsorbed H<sub>2</sub>O increases and the substrate feature again stays almost constant. The steady intensity of the O1s signal of the alumina substrate despite the increasing total adsorbate (H<sub>2</sub>O+NO<sub>2</sub>) coverage is particularly interesting. This result is consistent with an interpretation that in the early stages of the H<sub>2</sub>O deposition, water forms a uniform layer on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface and forces the NO<sub>2</sub> multilayer to form large 3D clusters. Under these conditions, the photoelectron yield from the oxygen ions in alumina substrate is somewhat higher than that when a



thick NO<sub>2</sub> multilayer covers the  $\theta$ -Al<sub>2</sub>O<sub>3</sub> surface evenly. Naturally, as the water film thickness further increases (3rd and 4th H<sub>2</sub>O dose), the XPS signal from the water layer increases significantly, eventually yielding a clearly identifiable feature at 533.1 eV, while that of the substrate decreases due to the attenuation originating from the thick water layer. The results of the XPS studies on NO<sub>2</sub> and H<sub>2</sub>O multilayers on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface given here are also consistent with the current TPD investigations on similar coadsorption systems (Figure 5), suggesting the formation of 3D NO<sub>2</sub> clusters on top of the H<sub>2</sub>O layer in contact with  $\theta$ -Al<sub>2</sub>O<sub>3</sub> thin film.

#### 4. Conclusions

In this work we examined the interaction of H<sub>2</sub>O and NO<sub>2</sub> molecules on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) model catalyst surface using TPD, IRAS, and XPS techniques. Our findings can be summarized as follows:

(a) Based on the TPD experiments on H<sub>2</sub>O/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) and NO<sub>2</sub>/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) systems, desorption energies for H<sub>2</sub>O and NO<sub>2</sub> molecules were estimated to be 44.8 kJ/mol and 36.6 kJ/mol, respectively.

(b) IRAS results revealed that the NO<sub>2</sub> dimerizes at high surface coverages on the clean  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface and forms condensed D<sub>2h</sub>-N<sub>2</sub>O<sub>4</sub> (O<sub>2</sub>N–NO<sub>2</sub>) layers where the N–N bond is oriented parallel to the alumina surface. The presence of an underlying H<sub>2</sub>O multilayer was found to alter orientations of some of the N<sub>2</sub>O<sub>4</sub> molecules that are probably located at the H<sub>2</sub>O/NO<sub>2</sub> interface.

(c) Various coverage and temperature-dependent adsorption regimes were observed in TPD experiments for NO<sub>2</sub>/H<sub>2</sub>O mixtures due to the competition for the adsorption sites on the  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface. In the first regime where  $\theta_{\text{H}_2\text{O}} < 0.3$  and  $0.2 \text{ ML} < \theta_{\text{NO}_2} < 1.2 \text{ ML}$ , NO<sub>2</sub>, and other NO<sub>x</sub> species exist in mixed and segregated domains by forming 2D (N<sub>2</sub>O<sub>4</sub>/NO<sub>2</sub>/NO<sub>3</sub><sup>−</sup>/NO<sub>2</sub><sup>−</sup>/H<sub>2</sub>O) islands and 3D (NO<sub>2</sub>/N<sub>2</sub>O<sub>4</sub>) clusters. In the second regime where  $\theta_{\text{H}_2\text{O}} > 0.3 \text{ ML}$ , H<sub>2</sub>O molecules start to dominate the surface by occupying the majority of the available  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) sites, thus forcing NO<sub>2</sub> molecules to form exclusively 3D clusters. At higher temperatures (such as  $T > 180 \text{ K}$ ), nitrate and nitrite species with a small surface concentration ( $< 0.4 \text{ ML}$ ) can form mixed domains with the H<sub>2</sub>O molecules, which leads to a slight stabilization of these adsorbed water molecules.

(d) For the NO<sub>2</sub> (multilayer)/H<sub>2</sub>O(10 ML)/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) system, changes in the crystal structure of the underlying H<sub>2</sub>O layers during the amorphous solid water  $\rightarrow$  crystalline ice transition decreases the solubility/entrapment of NO<sub>2</sub> in the ice matrix.

(e) Strongly bound ionic NO<sub>x</sub> species such as nitrites and nitrates are formed during TPD experiments on a H<sub>2</sub>O/ $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100) surface where all of the available alumina adsorption sites are blocked with 10 ML of preadsorbed H<sub>2</sub>O molecules.

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- (40) Temperature values given in this work are calibrated using the amorphous solid water  $\rightarrow$  crystalline ice phase transition for 5 and 12 ML H<sub>2</sub>O layers on  $\theta$ -Al<sub>2</sub>O<sub>3</sub>/NiAl(100).
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