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# Phase Equilibria in the C<sub>60</sub> + Ferrocene System and Solid-State Studies of the C<sub>60</sub>·2Ferrocene Solvate

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Triclinic C<sub>60</sub>·2ferrocene, which melts peritectically at 495 K ( $\Delta_{\pi}H = +50 \text{ J g}^{-1}$ ), can form by direct union of its components with a negative excess volume of  $-68 \text{ Å}^3$  per formula unit, although the enthalpy of deferrocenation into fcc C<sub>60</sub> and ferrocene vapor is virtually the same as for the sublimation of pure ferrocene. C<sub>60</sub> solubility in molten ferrocene is about 0.5 mol % at 495 K. The strong anisotropy of the thermal expansion tensor illustrates the anisotropy of intermolecular interactions inferred from the crystal structure in the 90–300 K range. The existence of nonnegligible interactions is corroborated by the persistence of the crystalline local order in the amorphous phase obtained by grinding, which reverts to the initial crystalline phase upon heating. Thus, from the thermodynamic and structural points of view, C<sub>60</sub>·2ferrocene behaves as the C<sub>60</sub>·2S<sub>8</sub> solvate.

## Introduction

Molecular disorder in pure face-centered cubic C<sub>60</sub> crystals<sup>1–6</sup> at room temperature is the sole example that almost complies with the Pauling-Fowler model<sup>7,8</sup> ac-

cording to which molecules freely rotate in their crystal-line sites. The space group changes through a first-order phase transition<sup>6</sup> from fcc  $Fm\bar{3}m$ <sup>9</sup> to simple cubic  $Pa\bar{3}$ <sup>10,11</sup> at 260 K. Within this new lattice, the molecules undergo orientational motions around two different directions.<sup>12</sup>

Whether and how these disorders are still present in C<sub>60</sub> solvates in which IR spectroscopy shows almost no,<sup>13</sup> weak,<sup>14</sup> or some<sup>15</sup> charge-transfer remains to be an-

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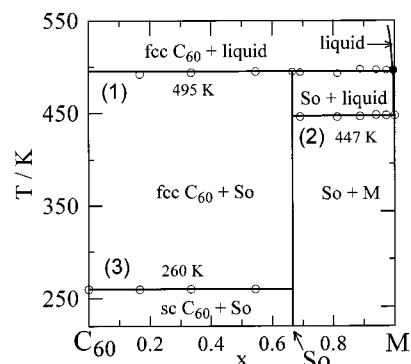
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swered. Unsuccessful attempts to locate each atom in the unit cells of these solvates at room temperature are related to large amplitude motions of  $C_{60}$  molecules that still persist in the solvate crystals.<sup>16–18</sup> However, a two-position orientational disorder of  $C_{60}$  was found to account for the crystal structure of the  $C_{60}$ ·4 benzene solvate at 104 K.<sup>19</sup>

Only two solvate structures, those of  $C_{60}$ ·2 $S_8$ <sup>20,21</sup> and  $C_{60}$ ·2ferrocene,<sup>22</sup> were solved at 298 K by determining and refining the coordinates of individual C-atoms, as if the  $C_{60}$  molecules were motionless. In addition, solid-state studies of  $C_{60}$ ·2 $S_8$ <sup>23,24</sup> resulted in rather paradoxical conclusions: although this solvate forms with a negative excess volume and local order persists after mechanical amorphization (both results indicating “strong” intermolecular interactions), sulfur was found to behave as if it were free from interactions with  $C_{60}$  molecules. Moreover, <sup>13</sup>C NMR studies showed that  $C_{60}$  molecules in the monoclinic  $C_{60}$ ·2 $S_8$  crystals undergo large amplitude motions around two or more axes.<sup>25</sup> Preliminary NMR experiments<sup>26–28</sup> at room temperature showed that the  $C_{60}$  molecules undergo fast re-orientational motions also within the  $C_{60}$ ·2ferrocene solvate and suggested that “the ferrocene mobility becomes much higher in  $C_{60}$ ·2ferrocene due to the extreme mobility of the surrounding fullerenes”. It was even said that “the change in ferrocene mobility ... in comparison with the pure ferrocene can be estimated ... as a factor of 70”.<sup>26</sup>

This paper presents solid-state studies on the  $C_{60}$  + ferrocene system and on the  $C_{60}$ ·2ferrocene solvate. The purpose was to characterize them and to compare the results with data previously reported on the  $C_{60}$  + sulfur system.<sup>23,24</sup> In particular, it had to be checked to determine whether the triclinic  $C_{60}$ ·2ferrocene solvate, crystallized from benzene solutions of both components at room temperature,<sup>22</sup> could form directly from solutions of  $C_{60}$  in molten ferrocene, and whether other solvates existed in the  $C_{60}$  + ferrocene system. In addition, it had to be observed for possible phase transitions at about 260 and 295 K in  $C_{60}$ ·2ferrocene, as suggested from previous DSC experiments,<sup>28</sup> al-



**Figure 1.**  $T$ - $x$  phase diagram of the  $C_{60}$  + ferrocene system. So = triclinic  $C_{60}$ ·2ferrocene, M = monoclinic ferrocene,  $x$  = mole fraction of ferrocene, ● = peritectic liquid.

though the crystal structure was found to be the same at 143 and 296 K.<sup>22</sup>

## Experimental Section

**Sample Preparation.** Fcc  $C_{60}$  purchased from TermUSA (purity > 99.98%) was used as such and ferrocene ( $C_{10}H_{10}Fe$ ) from Lancaster (purity: 98%) was sublimed before use.

One-hundred-milligram aliquots of  $C_{60}$ ·2ferrocene solvate were prepared by mixing both components in the  $C_{60}$ /ferrocene = 1/2 molar ratio in silica tubes sealed under a  $10^{-5}$ -Torr vacuum. These samples were heated to 458 K, annealed during 15 days at this temperature, and then slowly allowed to cool (during 2 days or so) to room temperature.

Samples for differential scanning calorimetry (DSC) studies were prepared by mixing different proportions of  $C_{60}$ ·2ferrocene and  $C_{60}$  or ferrocene in SETARAM high-pressure stainless steel pans (30- $\mu$ L inner volume) which were as fully filled as possible (the sample masses were about 23 mg) in order to reduce the sublimation of ferrocene to a minimum. These samples were annealed during 15 days at 423 K and slowly cooled to room temperature within a few days.

**Techniques.** High-resolution X-ray powder diffraction measurements (Debye–Scherrer geometry and transmission mode) were performed using the  $CuK\alpha_1$  ( $\lambda = 1.5406$  Å) radiation by means of two INEL CPS-120 diffractometers, working with 0.9 and 1.2 kW power, respectively. The latter was equipped with an OSMIC reflecting mirror in the incident beam and with an INEL cooling device working in the 90–300 K range. To prevent samples from possible shear stress-induced decomposition,<sup>29</sup> they underwent only gentle crushing before being introduced into 0.5-mm diameter Lindemann capillaries. For sample examinations long exposure time ranges (up to 12 h) were used because of X-ray absorption by Fe atoms.

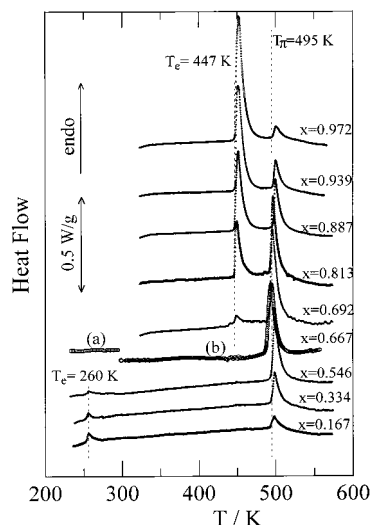
DSC and thermogravimetry (TG) were performed at a 5 K  $min^{-1}$  rate under a nitrogen flux with a DSC-10 cell and a TGA-50 balance of a TA2000 T.A.-Instruments thermal analyzer, respectively. Preliminary DSC experiments were performed in the 210–320 K range using a T.A.-Instruments cooling accessory. Sample masses were weighed by means of a 0.01-mg sensitive microbalance. A differential scanning calorimeter (MCB calorimeter, ARION) was also used for measuring sublimation enthalpies at a 1 K  $min^{-1}$  rate.

## Results

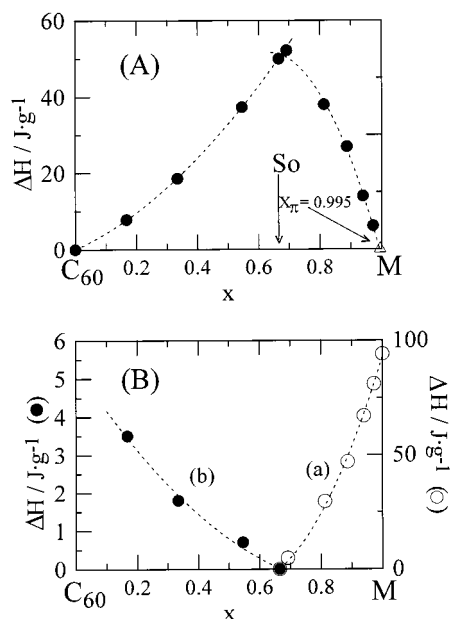
**Phase Diagram.** The  $T$ - $x$  phase diagram shown in Figure 1 was obtained by plotting the onset temperatures of the endothermic effects recorded on heating as a function of the ferrocene mole fraction ( $x$ ).

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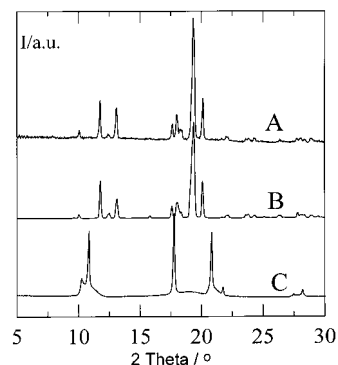
**Figure 2.** DSC curves on heating binary samples (23.3–24.8 mg) in stainless steel high-pressure pans (30- $\mu$ L inner volume). The bold curve corresponds to C<sub>60</sub>·2ferrocene: (a) heating from 213 to 303 K, (b) heating from 298 to 553 K.



**Figure 3.** Tammann curves. Enthalpy changes ( $\Delta H$ ) in J g<sup>-1</sup> as a function of mole fraction  $x$  for (A) peritectic equilibrium at 495 K ( $x_\pi$  = mole fraction of the peritectic liquid); (B) curve a = eutectic equilibrium at 447 K, curve b = eutectoid equilibrium at 260 K.

The endothermic peak observed at  $495 \pm 2$  K in the whole composition range (Figure 2), is assigned to the peritectic behavior of a binary compound. Its stoichiometry (C<sub>60</sub>/ferrocene = 1/2) is determined by drawing the Tammann curve (Figure 3A) associated with the peritectic equilibrium (horizontal line (1) in Figure 1). The composition of the peritectic liquid is found to be  $x_\pi = 0.995$  by extrapolating the right-hand part of the Tammann curve to  $\Delta H = 0$ . The enthalpy change  $\Delta_\pi H$  for the C<sub>60</sub>·2ferrocene peritectic melting was determined to be 50 J g<sup>-1</sup> (54.6 kJ mol<sup>-1</sup>,  $M(\text{C}_{60} \cdot 2\text{ferrocene}) = 1092$  g mol<sup>-1</sup>) at the crossing point of the two branches of the Tammann curve (Figure 3A).

The endothermic peak observed at  $447 \pm 1$  K in the  $0.666 < x = 1$  range (Figure 2) is assigned to a eutectic



**Figure 4.** Room-temperature high-resolution X-ray powder diffraction profiles. Experimental (A) and calculated (B) profiles for triclinic C<sub>60</sub>·2ferrocene. C = C<sub>60</sub> powder after deferrocenation of C<sub>60</sub>·2ferrocene.

equilibrium (line (2) in Figure 1) between the C<sub>60</sub>·2ferrocene solvate and pure ferrocene ( $T_{\text{fus}} = 447$  K<sup>30</sup>). An enthalpy change value of 92.3 J g<sup>-1</sup> was found by extrapolating the related Tammann curve (curve (a) in Figure 3B) to  $x = 1$ . This is very close to the value (94 J g<sup>-1</sup> at 447 K) for the melting enthalpy of pure ferrocene encapsulated in high-pressure DSC pans. These results account for the degeneracy of the eutectic equilibrium that precludes the solubility value at this temperature to be estimated. Now assuming that the ferrocene mole fraction of the eutectic liquid is at least equal to that of the peritectic liquid, a eutectic temperature of 446.5 K is calculated according to the Schröder equation.

Another endothermic effect occurs at  $260 \pm 1$  K in the C<sub>60</sub>-side part of the diagram (line (3) in Figure 1). It is assigned to a eutectoid equilibrium between C<sub>60</sub> and C<sub>60</sub>·2ferrocene. The related Tammann curve corresponds to curve (b) in Figure 3B. Extrapolating  $x$  to  $\Delta H = 0$ , a value of 0.667, i.e. the mole fraction for C<sub>60</sub>·2ferrocene, is found.

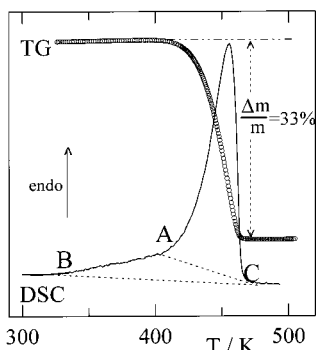
No effect, which would allow part of the liquidus curve to be drawn in the ferrocene-rich side of the diagram, was recorded at  $T > 495$  K. This indicates that the liquidus curve is probably very steep, as tentatively drawn in Figure 1.

**Thermal Stability and Deferrocenation of C<sub>60</sub>·2Ferrocene.** Powders of pure C<sub>60</sub>·2ferrocene synthesized by direct mixing of its components are made of grains whose shapes are reminiscent of the morphologies of the initial C<sub>60</sub> crystals. Nevertheless X-ray characterization and DSC experiments revealed no excess of either C<sub>60</sub> or ferrocene in the solvate powders. Their high-resolution X-ray diffraction profile at 298 K is virtually the same as the one calculated using the atom coordinates for triclinic C<sub>60</sub>·2ferrocene at room temperature,<sup>22</sup> as shown in Figure 4.

Deferrocenation process starts at about 400–420 K on heating at 5 K min<sup>-1</sup>. The one-step weight losses (30–36%) found from TG experiments (Figure 5) are close to the ideal value (34.1%). An enthalpy change of 120–150 J per gram of solvate (i.e. 65.5–82.0 J per mole of ferrocene) was found for deferrocenation by integrating the related endothermic effects within the range suggested by the TG curves. This is close to the

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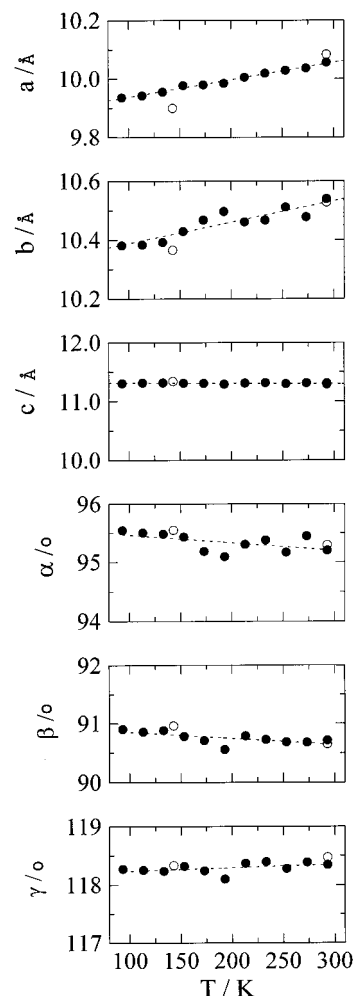
**Figure 5.** DSC and TG curves of  $C_{60}\cdot 2\text{ferrocene}$  in open pans. Sample masses: DSC = 5.84 mg, TG = 6.47 mg. Integrating the DSC peak within the A–C and B–C ranges gives 120 and 150  $\text{J g}^{-1}$ , respectively.

sublimation enthalpy values (64.6–84  $\text{kJ mol}^{-1}$ ) for pure ferrocene in the same temperature range.<sup>31–33</sup> To confirm this result, another experiment was performed, as for  $C_{60}\cdot 2S_8$ :<sup>23</sup> weighed amounts of  $C_{60}\cdot 2\text{ferrocene}$  were placed in sealed silica tubes ( $\approx 40$  cm long, 0.6 cm diameter) introduced vertically into one of the two cells of the MCB-ARION differential calorimeter. The main part of each tube ( $\approx 35$  cm) outside the cell remained at room temperature while the sample inside the cell was heated at a  $1 \text{ K min}^{-1}$  rate. The areas of the endothermic effects recorded while red-brown crystals of ferrocene grew in the middle part of the tubes were equal to the areas recorded on heating the same masses of pure ferrocene.

Room-temperature X-ray diffraction examination of powders after deferrocenation had occurred shows that the process is accompanied with the formation of the fcc  $C_{60}$  phase (profile C in Figure 4). This profile exhibits the same features as those recorded from other desolvated solvates<sup>34,35</sup> and related to defects and stacking faults.<sup>36</sup>

Because the crystal structures of  $C_{60}\cdot 2\text{ferrocene}$  at 143 K and room temperature are the same,<sup>22</sup> X-ray powder diffraction experiments were performed in the 90–293 K range by steps of about 20 K on heating from 90 K in order to determine thermal expansion and to look for possible (unlikely) phase transition(s). The lattice parameters were refined at each temperature (Figure 6) using the “pattern-matching” option of the FULLPROF program.<sup>37</sup> Then refined lattice parameters and unit-cell volume were fitted as a function of temperature by means of a standard least-squares method. Linear equations for these variations are gathered in Table 1.

To examine the anisotropy of the intermolecular interactions, the thermal-expansion tensor was deter-



**Figure 6.** Lattice parameters of triclinic  $C_{60}\cdot 2\text{ferrocene}$  as a function of the temperature. Empty circles correspond to values taken from the crystal structures at 143 and 296 K.<sup>22</sup>

**Table 1.** Linear Equations  $p = p_0 + p_1 T$  to Which Lattice Parameters Were Fitted

parameter	$P_0/\text{\AA}$	$p_1 \times 10^3/\text{\AA}\cdot\text{K}^{-1}$	$R^a \times 10^4$
$a$	9.878(4)	0.596(22)	4
$b$	10.316(22)	0.72(11)	20
$c$	11.313(7)	−0.024(4)	6
$\alpha$	95.6(1)	−1.3(6)	13
$\beta$	90.93(7)	−0.96(4)	7
$\gamma$	118.14(9)	0.46(4)	7

<sup>a</sup>  $R$ , defined as  $R = \sum (y_o - y_c)^2 / y_c^2$ , is the reliability factor between measured ( $y_o$ ) and calculated ( $y_c$ ) lattice constants.

mined by means of the DEFORM program.<sup>38</sup> The lattice deformation,  $dU$ , due to temperature variation,  $dT$ , is expressed in terms of a second-rank tensor  $dU_{ij} = \alpha_{ij} dT$ , where  $\alpha_{ij}$  ( $\text{K}^{-1}$ ) are the coefficients of the thermal-expansion tensor. Figure 7A shows the three-dimensional plot of the thermal-expansion tensor at 283 K in the frame of the principal axes  $\alpha_1$ – $\alpha_2$ – $\alpha_3$ , together with the crystallographic axes.

**Mechanical Grinding of Triclinic  $C_{60}\cdot 2\text{Ferrocene}$ .** For comparison with  $C_{60}\cdot 2S_8$  solvate,  $C_{60}\cdot 2\text{ferrocene}$  was ground at room temperature using pestle and mortar with increasing grinding times. X-ray powder profiles (Figure 8) of ground powders indicate that the crystal-

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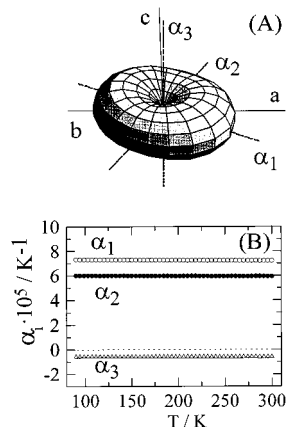
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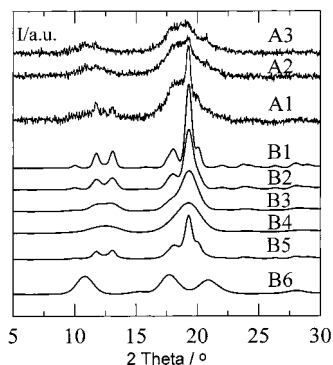
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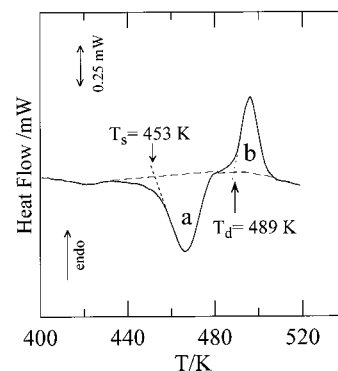


**Figure 7.** (A) Thermal-expansion tensor of triclinic C<sub>60</sub>·2ferrocene at 283 K in the frame of the principal directions  $\alpha_1$ ,  $\alpha_2$ ,  $\alpha_3$  (dashed lines) together with the crystallographic axes (continuous lines). The full length of the  $\alpha_i$  axes corresponds to  $2 \times 10^{-4} \text{ K}^{-1}$ . (B)  $\alpha_i$  ( $i = 1, 2, 3$ ) principal coefficients as a function of the temperature for triclinic C<sub>60</sub>·2ferrocene.



**Figure 8.** Experimental X-ray profiles of triclinic C<sub>60</sub>·2ferrocene as a function of the grinding time: A1 = 30 min, A2 = 60 min, A3 = 90 min. Calculated X-ray profiles of triclinic C<sub>60</sub>·2ferrocene as a function of the crystallite size: B1 = 150 Å, B2 = 100 Å, B3 = 60 Å, B4 = 40 Å, B5 = calculated profile for a mixture of 40-Å (30%) and 150-Å (70%) crystallite sizes, B6 = calculated profile for a mixture of fcc C<sub>60</sub> and monoclinic ferrocene with a 50-Å crystallite size.

line sample transforms into a noncrystalline (nc) phase within 90 min or so. The DSC curve (Figure 9) of the nc phase obtained after a 90-min grinding exhibits an exothermic effect (peak a) at 453 K (onset) associated with an enthalpy change of about  $-37.5 \text{ J g}^{-1}$  while heating at a  $5 \text{ K min}^{-1}$  rate. It is followed by an endothermic peak (b) at 489 K (onset) reminiscent of the peritectic melting of the triclinic solvate, as confirmed by the X-ray diffraction at room temperature of the sample recovered at the end of the run. It should be noticed that this profile revealed the presence of some fcc C<sub>60</sub> besides triclinic C<sub>60</sub>·2ferrocene. This indicates that the solvate partially dissociates into fcc C<sub>60</sub> and ferrocene vapor on heating. A possible explanation is that the sample volume ( $1.2 \mu\text{L}$ ) is small with respect to the inner volume ( $30 \mu\text{L}$ ) of the high-pressure DSC pan, which can then be saturated with the sample vapor (i.e. almost pure ferrocene). This may explain why the enthalpy value from peak b in Figure 9 is found to be half ( $25.3 \text{ J g}^{-1}$ ) that obtained when samples practically fill the inner volumes of the DSC pans ( $50 \text{ J g}^{-1}$ ). Indeed, an incomplete transformation of the non crystalline phase might also explain these results.



**Figure 9.** DSC curve obtained on heating a C<sub>60</sub>·2ferrocene sample previously subjected to a 90-min grinding.

## Discussion

The C<sub>60</sub>·2ferrocene solvate, the sole molecular binary compound found in the C<sub>60</sub> + ferrocene system, was shown to form from molten ferrocene, as C<sub>60</sub>·2S<sub>8</sub> does from molten sulfur,<sup>23</sup> although the C<sub>60</sub> solubility in both components is very low (ca. 0.005 mole per mole of solution at 450 K).

The volume of the triclinic C<sub>60</sub>·2ferrocene at 296 K is  $1048 \text{ Å}^3$  per formula unit.<sup>22</sup> Assuming a molecular volume of  $710 \text{ Å}^3$  per C<sub>60</sub> molecule, as in the fcc unit-cell ( $a = 14.16 \text{ Å}$ ), about  $338 \text{ Å}^3$  remain available for the two ferrocene molecules. If it is assumed that a ferrocene molecule occupies a volume of  $203 \text{ Å}^3$  per ferrocene molecule, as it does in the ferrocene monoclinic unit-cell,<sup>39</sup> the C<sub>60</sub>·2ferrocene triclinic solvate forms with a negative excess volume of  $-68 \text{ Å}^3$  per formula unit.

Virtually the same excess volume is found at 143 K taking  $V/Z = 1017 \text{ Å}^3$ <sup>22</sup> for triclinic C<sub>60</sub>·2ferrocene,  $V = 698 \text{ Å}^3$  per C<sub>60</sub> molecule, as in the sc unit-cell ( $a = 14.08 \text{ Å}$  at 143 K), and  $V/Z = 193 \text{ Å}^3$  and  $196 \text{ Å}^3$  at 149 K for orthorhombic and triclinic ferrocene, respectively.<sup>40,41</sup> A linear extrapolation of the ferrocene available volumes ( $338 \text{ Å}^3$  at 296 K and  $319 \text{ Å}^3$  at 143 K per 2 molecules) to 0 K gives a  $301 \text{ Å}^3$  volume that hits the calculated van der Waals volume ( $2 \times 149 \text{ Å}^3$ ) for two ferrocene molecules. Now assuming a van der Waals volume of  $524 \text{ Å}^3$  for a C<sub>60</sub> "sphere" with 5 Å radius, the packing coefficient for C<sub>60</sub>·2ferrocene is 0.785 at 296 K and 0.808 at 143 K. Linearly extrapolating the unit-cell volume of C<sub>60</sub>·2ferrocene to 0 K gives  $988 \text{ Å}^3$  and a packing coefficient as high as 0.83. Moreover calculating the volume occupied in the triclinic unit-cell by a C<sub>60</sub> molecule at 0 K (i.e. by subtracting ( $298$  or  $301 \text{ Å}^3$ ) from  $988 \text{ Å}^3$ ), a value of  $687$  or  $690 \text{ Å}^3$  is found, which is the close to that extrapolated to 0 K for the hexagonal polymorph of C<sub>60</sub> ( $682 \text{ Å}^3$ ).<sup>42</sup>

To sum up, a negative excess volume and a high packing coefficient allow "strong" interactions between C<sub>60</sub> and ferrocene molecules to be inferred from crystallographic data, as found previously for a number of C<sub>60</sub> solvates.<sup>43</sup>

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The molecular packing in triclinic  $C_{60} \cdot 2\text{ferrocene}$  ( $a = 10.084 \text{ \AA}$ ,  $b = 10.528 \text{ \AA}$ ,  $c = 11.306 \text{ \AA}$ ,  $\alpha = 95.29^\circ$ ,  $\beta = 90.65^\circ$ ,  $\gamma = 118.47^\circ$  at  $296 \text{ K}^{22}$ ), made of alternating  $C_{60}$  and ferrocene layers, may be considered as resulting from a distortion of a simple hexagonal packing like that in  $C_{60} \cdot x \text{I}_2^{44}$  or  $C_{60} \cdot 2 \text{CCl}_4^{45}$ . Thus, ferrocene molecules are intercalated between dense  $C_{60}$  planes at an angle with respect to their own reticular planes. Moreover, the 5-fold axes of the ferrocene molecules are nearly parallel to direction  $u \ v \ w = 3 \ 6 \ 4$ , as one 5-fold axis of every  $C_{60}$  molecule is.

In the triclinic lattice,  $2 \times 5$  shortest  $C \cdots H(-C)$  distances ( $< 3.1 \text{ \AA}$ ), i.e. the "strongest" intermolecular interactions, are found to link any  $C_{60}$  molecule and its surrounding ferrocene molecules nearly along the  $0 \ 1 \ 1$  direction, thus resulting in an increased values of angle  $\alpha$  with respect to the ideal hexagonal packing. Nevertheless  $2 \times 16$  supplementary  $C \cdots H(-C)$  distances in the  $3.1\text{--}3.5 \text{ \AA}$  range yield interactions which, as a whole, are roughly perpendicular to  $C_{60}$  and ferrocene layers.

This is illustrated by the shape of the thermal expansion tensor (Figure 7A) whose  $\alpha_3$  direction, the "hard direction" along which strong interactions operate, is nearly perpendicular to molecular layers, i.e. nearly parallel to axis  $c$ . In addition, the "soft"  $\alpha_1$  and  $\alpha_2$  directions, whose coefficients are close to each other, lie in a plane parallel to plane  $ab$ , i.e. to homo-molecular layers. Data in Figure 7B show that this picture of the  $C_{60}$ -ferrocene interactions does not depend on temperatures in the  $90\text{--}300 \text{ K}$  range.

Transformation of triclinic  $C_{60} \cdot 2\text{ferrocene}$  into a non-crystalline phase upon grinding further supports the previous inferences for the existence of "strong" interactions between  $C_{60}$  and ferrocene molecules. The experimental profiles recorded after increasing grinding-time ranges were compared with those calculated by decreasing the crystallite size to  $40 \text{ \AA}$  (Figure 8). It can be seen that the 90-min profile is similar to the  $40\text{-\AA}$  profile, although a small contribution of the mixture of pure components with a  $50\text{-\AA}$  particle size (profile B6 in Figure 8) cannot be discarded. In addition, the 30-min profile can result from the contribution of various crystallite sizes. Although the structure does not survive grinding, as proven by the exothermic recrystallization of the amorphous phase (peak a in Figure 9), the agreement between experimental and calculated profiles

indicates that the local order of the triclinic phase is retained in the noncrystalline phase and resists grinding.

Nevertheless, the deferrocenation enthalpy was found to be equal to the sublimation enthalpy of pure ferrocene, thus suggesting that no extra-interaction between  $C_{60}$  and solvating molecules takes place on forming solvates.

### Concluding Remarks

The phase-diagram method brings basic and reliable information to the binary compounds it reveals. As far as  $C_{60} \cdot 2\text{ferrocene}$  is concerned, it showed that no crystal-crystal phase transition occurs in the  $90\text{--}495 \text{ K}$  range. This disagrees with previous DSC measurements which revealed two endothermic peaks at  $260$  and  $295 \text{ K}$  (onsets), respectively,<sup>28</sup> although the crystal structures were found to be the same at  $143 \text{ K}$  and at  $296 \text{ K}^{22}$ . Thus the chemical identity of the sample previously examined through Mossbauer spectroscopy remains elusive.

Although  $^{13}\text{C}$  NMR studies have shown that intermolecular interactions do not prevent  $C_{60}$  molecules from undergoing large amplitude motions within the monoclinic  $C_{60} \cdot 2\text{S}_8$  crystals, it was possible to ascribe their motionless appearance, from the crystallographic point of view, to jumps between equivalent (symmetry-related) molecular orientations, probably because molecular symmetry axes are very close to reticular axes.<sup>25</sup> In the case of  $C_{60} \cdot 2\text{ferrocene}$ , NMR studies have already shown that the  $C_{60}$  molecules undergo fast reorientational motions, whereas the ferrocene molecules move more slowly if they move at all.<sup>26</sup> Future dynamic studies should answer this question. Another question to be answered arises from the near parallelism between the 5-fold axes of ferrocene molecules and a 5-fold axis of every  $C_{60}$  molecule. It remains to be established whether there is some coupling between the possible motions of both kinds of molecules around their respective 5-fold axes: NMR studies as a function of temperature should shed some light on this problem.

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