Vibrational Spectra of C₆₀·C₈H₈ and C₇₀·C₈H₈ in the Rotor-stator and Polymer Phases

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 $C_{60} \cdot C_8 H_8$ and $C_{70} \cdot C_8 H_8$ are prototypes of rotor-stator cocrystals. We present infrared and Raman spectra of these materials and show how the rotor-stator nature is reflected in their vibrational properties. We measured the vibrational spectra of the polymer phases poly($C_{60}C_8 H_8$) and poly($C_{70}C_8 H_8$) resulting from a solid-state reaction occurring on heating. On the basis of the spectra, we propose a connection pattern for the fullerene in poly($C_{60}C_8 H_8$), where the symmetry of the C_{60} molecule is D_{2h} . On illuminating the $C_{60} \cdot C_8 H_8$ cocrystal with green or blue light, a photochemical reaction was observed leading to a product similar to that of the thermal polymerization.

1. Introduction

Fullerenes and cubane have recently been shown to form so-called rotor-stator cocrystals. These cocrystals are different from both orientationally ordered and plastic crystals, as one of their constituents (the fullerene) is rotating and the other one (the cubane) is fixed in a well-defined orientation. In the case of $C_{60} \cdot C_8 H_8$, rotating C_{60} molecules form a face centered cubic lattice and static cubane molecules, occupying interstitial octahedral sites, serve as bearings between them. $C_{70} \cdot C_8 H_8$ crystallizes in a face-centered cubic structure above 375 K. At room temperature, the rotation of C_{70} is somewhat restricted, which leads to a tetragonal distortion; the C_{70} molecule is able to rotate around its main axis which, in turn, precesses around the crystallographic c axis. The formation of these structures is driven by the molecular recognition between the concave surface of the cubane and the round surface of the fullerenes. 1,3

On heating, the fullerene—cubane compounds undergo a topochemical reaction.¹ As the reaction product is insoluble in common solvents, it is most likely a copolymer of the fullerene with cubane.⁴ X-ray diffraction patterns of the annealed samples, measured at room temperature, show a large emerging amorphous part and weakening reflections compatible with fcc structure. Compared with the original monomer phase, the shift of these reflections indicates lattice expansion, and their intensity quickly vanishes at high angles. Because of the parallel appearance of the amorphous contribution and disappearance of crystallinity, we can assume that the amorphous phase retains the local cubic order. Another observation which makes this assumption reasonable is that the morphology of the crystals does not change on heating.¹

In this paper, we present a detailed vibrational (infrared and Raman) characterization of the monomer and polymer phases of $C_{60} \cdot C_8 H_8$ and $C_{70} \cdot C_8 H_8$. In the monomer phases, we can

confirm the rotor-stator nature of the materials. Based on the spectra of the polymer phases, we deduce the symmetry of the majority of the fullerene units as D_{2h} , similar to the linear cycloaddition polymers. This conclusion is consistent with a substantial presence of linear segments in the copolymer.

We published the infrared spectra of the monomer and polymer phases of $C_{60} \cdot C_8 H_8$ and $C_{70} \cdot C_8 H_8$ earlier as Supplementary Information to ref 1. A thorough study of the polymerization of $poly(C_{60}C_8H_8)$ at high temperature and high pressure has been performed by Iwasiewicz-Wabnig et al.,⁵ using X-ray diffraction and Raman spectroscopy. Our results, obtained at ambient pressure on annealing, are complementary to that study, except that we observe a photopolymerization reaction on illumination with green or blue light, which accounts for the laser wavelength dependence of the Raman spectra.

2. Experimental Methods

Cubane was prepared following the method of Eaton and Cole.⁶ Cubane and the fullerenes C_{60} and C_{70} were coprecipitated from toluene by adding isopropyl alcohol or by evaporating the solvent to form $C_{60} \cdot C_8 H_8$ and $C_{70} \cdot C_8 H_8$.¹

The resulting black powder was pressed into KBr pellets for infrared (IR) measurements. The spectra were recorded by a Bruker IFS28 and a Bruker IFS 66v/S spectrometer. Depending on the width of the lines to be observed, the resolution was set between 2 and 0.25 cm⁻¹. Temperature-dependent measurements were conducted in a flow cryostat cooled by liquid nitrogen or helium with the temperature adjustable between 20 and 600 K. The KBr pellet technique has the disadvantage that the index of refraction of the samples is generally in mismatch with that of the medium; therefore, the lineshapes become asymmetric (Christiansen effect). However, the alternative of using organic oils as Nujol was discarded because we wanted to identify as many new infrared lines as possible, without disturbing absorption from the medium.

Raman microscopy data were acquired in backscattering geometry on powder samples either under ambient conditions or in an evacuated glass capillary. Spectra were taken with three

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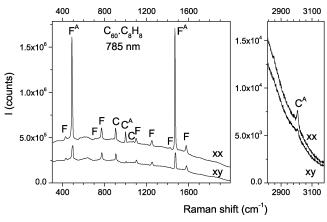


Figure 1. Room-temperature Raman spectra of the C_{60} · C_8H_8 cocrystal. The diode laser was operated at the line indicated. Spectra taken with the incident and scattered light polarizations parallel and perpendicular are labeled by xx and xy, respectively. Cubane modes⁷ are denoted by C, and fullerene modes⁸ are denoted by F. Totally symmetric modes are marked by superscript A.

lines (468, 531, and 676 nm) of a Kr-ion laser on a triple monochromator (Jobin-Yvon T64000). The laser power was carefully adjusted not to cause polymerization or any other type of changes in the samples. This was guaranteed with a power of $70-100~\mu\mathrm{W}$ focused to a spot of approximately 2 $\mu\mathrm{m}$ diameter. The slit width was set at 300 or 400 $\mu\mathrm{m}$. For these small crystals (typically less than $10~\mu\mathrm{m}$), the orientation of the principal axes with respect to the polarization of the incident (\mathbf{e}_i) and the scattered (\mathbf{e}_s) light could not be determined. However, in case of highly symmetric molecules, the fully symmetric A_g vibrations can easily be identified by comparing polarized ($\mathbf{e}_s \mid \mid \mathbf{e}_i$) and depolarized ($\mathbf{e}_s \perp \mathbf{e}_i$) spectra. For simplicity, we label these as xx and xy, respectively. The Raman spectra taken with the 785 nm laser line of a diode laser were collected by a Renishaw 1000 MB Raman spectrometer.

3. Results and Discussion

3.1. Rotor-Stator Phases. The Raman and infrared spectra of C₆₀•C₈H₈ in the rotor-stator phase are shown in Figures 1–3, and those of C₇₀•C₈H₈ are shown in Figures 4 and 5. The frequencies of the observed vibrational peaks of C₆₀•C₈H₈ are listed in Tables 1 and 2, and those of C₇₀•C₈H₈ are in Tables 3 and 4. We compare these frequencies to experimental data on cubane⁷ and C₆₀ (ref 8) and calculated Raman⁹ and infrared¹⁰ spectra of C₇₀, respectively. As expected for molecular cocrystals with the lattice stabilized by van der Waals interaction only, the spectra are superpositions of those of the constituents. As no crystal field splitting of the fullerene lines is observed, the infrared measurement confirms that the fullerene molecules are rotating in the crystal. The cubane lines are not split either, proving that the crystal field around the cubane has the same point group, that is, O_h , as that of the isolated molecule. In the Raman spectrum of the rotor-stator cocrystals taken with 785 nm excitation, the fullerene lines are significantly stronger than the cubane lines, most probably because of the enhanced Raman cross section caused by the conjugated bonds, similar to what was found in fullerene clathrates.11 This effect renders the cubane lines almost unnoticeable. When changing the wavelength of the exciting laser to 531 nm, all of the cubane lines are lost (Figure 2), because we approach resonant scattering in the fullerenes. 12

 C_{60} belongs to the icosahedral (I_h) point group and consequently shows four infrared-active vibrational modes with T_{1u}

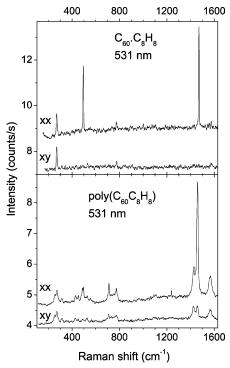


Figure 2. Raman spectra of C_{60} $\cdot C_8H_8$ at room temperature before annealing (monomer) and after annealing at 470 K (polymer). The Kr⁺ laser line and the polarizations are indicated. The spectra are vertically shifted for clarity.

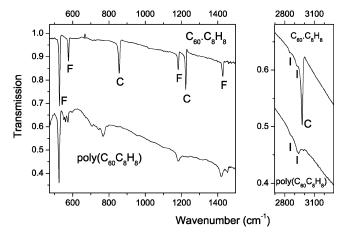


Figure 3. Infrared spectra of C_{60} • C_8H_8 before (cocrystal) and after annealing at 470 K (copolymer). C stands for cubane modes; F stands for fullerene modes, and I stands for impurity. The spectra are vertically shifted for clarity. The changes in the spectra show that annealing leads to the polymerization of the sample.

symmetry. Out of its 10 Raman-active modes, 2 belong to the A_g , and 8 belong to the H_g irreducible representation. We could observe all of these modes in the spectrum of $C_{60} \cdot C_8 H_8$ (the H_g (1) mode can be seen in Figure 2). C_{70} has D_{5h} symmetry and altogether 31 IR active and 53 Raman active vibrational modes. The IR modes can be decomposed as $21 E_1' + 10 A_2''$, and the Raman modes can be decomposed as $12 A_1' + 22 E_2' + 19 E_1''$. Similar to the case of pristine C_{70} , not all of these modes have sufficient intensity to be easily detected. Cubane belongs to the octahedral (O_h) point group. Its three infrared-active T_{1u} modes are clearly visible in the spectra of the $C_{60} \cdot C_8 H_8$ and $C_{70} \cdot C_8 H_8$ rotor-stator cocrystals. This cubane spectrum is indeed the closest to that of isolated cubane in a crystalline environment; solid cubane shows a more complicated spectrum because of the lower site symmetry. The 8 Raman-active modes

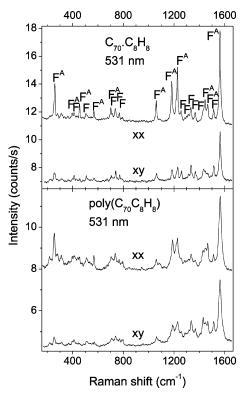


Figure 4. Room-temperature Raman spectra of C₇₀•C₈H₈ cocrystal and copolymer. The Kr⁺ laser line and the polarizations are indicated. The spectra are vertically shifted for clarity. Totally symmetric modes are denoted by superscript A.9 Fullerene peaks are marked by F;8 no cubane peaks were found.

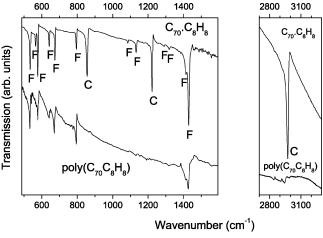


Figure 5. Infrared spectra of C₇₀•C₈H₈ before and after annealing at 470 K (cocrystal and copolymer phase, respectively). C: cubane peaks.⁷ F: fullerene peaks.8 The asymmetric line shape is due to the Christiansen effect.

of cubane are classified as $2A_{1g} + 2E_g + 4T_{2g}$. Only three out of these eight appear in the C₆₀•C₈H₈ spectrum taken with the 785 nm laser, and none appear in the spectra taken with the 531 nm laser, because of the aforementioned cross-section differences.

In the C₆₀•C₈H₈ cocrystal, the depolarization ratio $\rho = \phi_{xy}/\phi_{xx}$ (with ϕ_{ij} the oscillator strength of an excitation at either xy or xx polarization; see section 2) should be zero for the fullerene A_g modes and 3/4 for the H_g modes. The A_g modes were indeed found totally polarized, and the depolarization ratio was 0.90 for the H_g (1) and 0.71 for the H_g (4) mode (see Figure 2). In contrast, the totally symmetric modes of C_{70} should not vanish completely in the xy geometry because of its D_{5h}

TABLE 1: Raman Frequencies of the C₆₀·C₈H₈ Cocrystal and Poly(C₆₀C₈H₈) Copolymer, and Assignment^{7,8} of the Cocrystal Peaks Where C Stands for Cubane and F Stands for Fullerene Peaks

C_{60} • C_8H_8		$poly(C_{60}C_8H_8)$
$v^* \text{ (cm}^{-1})$	assignment	$v^* \text{ (cm}^{-1})$
271	$F, H_g(1)$	255
		272
		314
428	$F, H_g(2)$	429
		451
495	$F, A_g(1)$	486
		524
		560
708	$F, H_g(3)$	711
		732
	- · · · · ·	752
770	$F, H_g(4)$	774
904	C, E_g	
1000	C, A_{1g}	
1072	C, E_g	
1099	$F, H_g(5)$	
1248	$F, H_g(6)$	1.406
1423	$F, H_g(7)$	1426
1469	$F, A_g(2)$	1459
1576 3008	$F, H_g(8)$ C, A_{1g}	1566

TABLE 2: Infrared Frequencies of the $C_{60}{}^{\bullet}C_8H_8$ Cocrystal and $Poly(C_{60}C_8H_8)$ Copolymer and $Assignment^{7,8}$ of the Cocrystal Peaks Where C Stands for Cubane and F for **Fullerene Peaks**

C_{60} • C_8H_8		$poly(C_{60}C_8H_8)$
$v^* (cm^{-1})$	assignment	$v^* (cm^{-1})$
527	F, $T_{1u}(1)$	526
		551
		561
577	F, $T_{1u}(2)$	574
		705
		723
		742
		768
857	C, T_{1u}	
1181	$F, T_{1u}(3)$	1181
1224	C, T_{1u}	
1428	$F, T_{1u}(4)$	1424
		1458
2976	C, T_{1u}	2948

symmetry. This is what is found in the C₇₀•C₈H₈ cocrystal. The modes that have lower depolarization ratios are labeled by A in Figure 4. These modes correspond to the ones assigned to A'₁ by Sun and Kertész.⁹

In contrast to the fullerenes, the frequencies of the cubane principal lines in the rotor-stator crystals deviate from those of cubane in its pure solid form. If we compare the vibrational frequencies for various environments of the cubane molecule, a clear trend can be observed. The highest vibrational frequencies occur in the gas phase. 13 In pure solid cubane or in solution, the lines shift to lower frequencies. Further downshift is found in C₆₀•C₈H₈ and finally in C₇₀•C₈H₈. This trend is similar to that found in the vibrational frequencies of molecules trapped in rare gas matrices14 and is caused by van der Waals interaction: the higher the polarizability of the environment, the lower the frequency. The relatively large shifts in the solids reflect the high polarizability of the fullerenes.

3.2. Poly($C_{60}C_8H_8$). The spectra of $C_{60}\cdot C_8H_8$ change dramatically upon annealing to 470 K either in a furnace or in a heated cryostat in the IR spectrometer (Figure 3). The Raman and IR spectra of the annealed sample are plotted in Figures 2 and 3,

TABLE 3: Raman Frequencies of the $C_{70} \cdot C_8 H_8$ Cocrystal and Their Assignment According to Ref 9; All Peaks Are Fullerene Peaks, and the Peaks of Poly($C_{70}C_8H_8$) Have Essentially the Same Center Frequencies

v^* (cm ⁻¹)	assignment9
259	A_1'
397	$A_1^{\frac{1}{2}}$
411	$E_1^{''}$
454	A_1^{\dagger}
507	$A_1^{\prime} \ E_2^{\prime}$
568	$A_1^{\overline{\prime}}$
701	A_1^{\prime}
713	A_1' E_1'' E_1'' E_2' A_1' A_1' A_1' E_2' E_1'' E_2' E_1'' E_2' E_1'' E_1'' E_1'' E_1'' E_1'' E_1''
737	$E_1^{\prime\prime}$
769	E_2^{\prime}
1060	$A_1^{\overline{\prime}}$
1182	A_1'
1227	A_1'
1256	E_2'
1313	$E_1^{\prime\prime}$
1333	E_2'
1368	$E_1^{\prime\prime}$
1433	$E_1^{\prime\prime}$
1445	A_1'
1466	$A_1^{\prime} \ E_1^{\prime\prime}$
1512	$E_1^{\prime\prime}$
1564	A_1^{\prime}

TABLE 4: Infrared Frequencies of the $C_{70} \cdot C_8 H_8$ Cocrystal and $Poly(C_{70}C_8 H_8)$, and the Assignment of the Former According to Ref 10; C Stands for Cubane Peaks, and F Stands for Fullerene Peaks

C ₇₀ *C ₈ H ₈		poly(C ₇₀ C ₈ H ₈)
$v^* \text{ (cm}^{-1})$	assignment ¹⁰	$v^* \text{ (cm}^{-1})$
535	F, <i>E</i> ' ₁	533
	1	541
565	$F, A_2^{\prime\prime}$	565
		569
578	F, E'_1	578
		582
642	F, E'_1	641
		647
674	F, E'_1	671
		676
		763
795	F, E'_1	776
		794
856	C, T_{1u}	
1085	F, E'_1	1086
1133	$F, A_2^{\prime\prime}$	1132
		1154
		1190
		1202
		1217
1202	$F, A_2^{\prime\prime}$	
1222	C, T_{1u}	
1291	F, E'_1	
1319	F, A_2''	
1413	$F, E_1^{\overline{\prime}}$	1413
1429	F, E'_1	1427
2974	C, T_{1u}	2964

and the peak positions are listed in Tables 1 and 2, respectively. Upon heating to 470 K, an irreversible reaction takes place. When annealing a few tens of mg sample in the furnace, the first changes in the IR spectra appear after 40 min: C_{60} modes split, and new modes appear. Further annealing leads to the disappearance of the original C_{60} and cubane modes and increased intensity of the new peaks. The new features of the final reaction product in the IR spectrum are the same, irrespective of whether the annealing was done in a furnace or in situ in a cryostat.

In the Raman spectrum of the annealed $C_{60} \cdot C_8 H_8$, the A_g modes of C_{60} do not split, but the low energy, that is, radial H_{ϱ} modes show at least a threefold splitting, best seen on the lonestanding H_g (1) mode. In the IR spectrum, the original T_{1u} modes of the fullerene split into at least two lines, and new peaks appear between 700 and 800 cm⁻¹. The splitting and the new modes indicate that the C₆₀ molecule is distorted. However, the number of new lines is considerably less than what would be expected if the cage opened. 15 In contrast, the change in the cubane lines is striking. The original lines disappear completely; only a weak IR line at 2948 cm⁻¹ indicates that there are still hydrocarbon groups in the sample. We infer from the position of this line, which corresponds to the C-H stretching in saturated hydrocarbons, that the carbon atoms involved are sp³ hybridized. In the reaction, we have to account for all atoms since no mass loss was observed by thermogravimetry-mass spectrometry (TG-MS) up to 570 K.1 This suggests that the cubane transforms into a different constitutional isomer and covalently bonds to C_{60} , leading to a structural distortion. The reaction product is most probably a covalently bound copolymer, as the products are insoluble in common solvents.

Pristine cubane also isomerizes at 470 K, 16 the same temperature where the polymerization appears in $C_{60} \cdot C_8 H_8$. Hence, a straightforward assumption is that the first step of the copolymerization reaction must be the decomposition of cubane. Pristine cubane can decompose into several products, for example, cyclooctatetraene, bicyclooctatriene, styrene, and dihydropentalene. 16 As the first three form known adducts with C_{60} , 17 which we could not detect by either IR spectroscopy or HPLC, 4 we can exclude these as being the connecting units between the fullerenes.

In principle both fullerene-fullerene and fullerene-C₈H₈ bonds can be realized in the polymer. C₈H₈-C₈H₈ bonds can be excluded, as the C₈H₈ molecules are well-separated by the fullerene molecules. We can also exclude the possibility of covalent fullerene-fullerene bonding because of the following experimental observations. There are two known bond types between fullerene molecules in fullerene homopolymers. In neutral polymers, the [2 + 2] cycloaddition leads to a cyclobutane-type ring with two single bonds between the buckyballs. 18,19 A Raman peak at approximately 950 cm⁻¹ is associated with this bond.²⁰ This peak is absent in the spectrum of poly-(C₆₀C₈H₈). The other possible bond type is one single bond between two fullerene molecules.²¹ This bond leads to the appearance of a characteristic IR peak between 800 and 850 cm⁻¹. As this peak is also absent, we can rule out the fullerenefullerene direct bond. There is still another observation which confirms this assumption. In fullerene polymers^{22,23} and in the dimer-oxide C₁₂₀O,^{23,24} interball vibrational peaks appear in the Raman spectrum between 100 and 140 cm⁻¹. We measured the Raman spectrum down to 20 cm⁻¹ but did not find any peaks below the split $H_o(1)$ mode. The reason for the absence of the interfullerene bonding comes from structural considerations. The large interfullerene distance observed by X-ray diffraction¹ does not allow the C₆₀ molecules to approach each other close enough for a reaction to occur between them.

In the following, we try to establish the connection pattern of the fullerene unit based on the infrared and Raman spectra. Since the IR and Raman spectra retain mutual exclusion (no lines are observed to appear simultaneously in both), the inversion center of the C_{60} balls must be preserved. This means that the possible point groups of the 60 molecules are: I_h , T_h , S_6 , D_{5d} , D_{3d} , D_{2h} , C_{2h} , or C_i . In Table 5, we show the evolution and splitting of the Raman active A_g and H_g and the IR active

TABLE 5: Correlation Tables for the A_g , H_g , and T_{1u} Representations of I_h , for the Subgroups of I_h Containing Inversion Where R Denotes Raman and IR Denotes Infrared **Active Modes**

I_h	$A_g(\mathbf{R})$	$H_{g}\left(\mathbb{R}\right)$	$T_{1u}(IR)$
T_h	$A_g(\mathbf{R})$	$T_g(\mathbf{R}) + E_g(\mathbf{R})$	T_u (IR)
S_6	$A_g(\mathbf{R})$	$A_g(\mathbf{R}) + 2E_g(\mathbf{R})$	A_u (IR) + E_u (IR)
D_{5d}	$A_{1g}(\mathbf{R})$	$A_{1g}(R) + E_{1g}(R) + E_{2g}(R)$	A_{2u} (IR) $+ E_{1u}$ (IR)
D_{3d}	$A_{1g}\left(\mathbf{R}\right)$	$A_{1g}(\mathbf{R}) + 2 E_g(\mathbf{R})$	A_{2u} (IR) $+ E_u$ (IR)
D_{2h}	$A_g(\mathbf{R})$	$2 A_g(R) + B_{1g}(R) +$	B_{1u} (IR)+ B_{2u} (IR) +
		$B_{2g}(R) + B_{3g}(R)$	B_{3u} (IR)
C_{2h}	$A_g(\mathbf{R})$	$3 A_g(R) + 2 B_g(R)$	A_u (IR) $+2 B_u$ (IR)
C_i	$A_g(\mathbf{R})$	$5 A_g (R)$	$3 A_u$ (IR)

 T_{1u} modes caused by symmetry reduction from I_h to these point groups (correlation table). The C_{2h} and C_i point groups can be ruled out because the expected number of additionally activated peaks^{25,26} is too high to be reconciled with the observed data. A D_{2h} distortion could in principle be positively identified as it leads to a threefold splitting of the T_{1u} modes, in contrast to the others; unfortunately, in this case, our fits were not sufficiently robust to distinguish between a three- or twofold splitting. I_h or T_h symmetry would not cause splittings; therefore, these cannot be the only point groups appearing; there must be units of reduced symmetry even if the connection pattern of the fullerene units is not uniform throughout the whole polymer.

To draw the possible structures with the appropriate point groups, we recall our assumption based on structural data^{1,5} that the local arrangement of the molecules does not change significantly on polymerization; thus, the fullerenes must still be surrounded octahedrally by cubanes. In addition, on polymerization, the inversion center of the C₆₀ molecule can be retained only if it is connected to an even number of C₈H₈ molecules. The connection patterns selected by this condition from the set of possible point groups are depicted in Figure 6. This subset contains T_h , S_6 , D_{3d} , and D_{2h} .

Three types of fullerene-C₈H₈ connections appear in the possible structures. In the first case (patterns a, b, and d in the second column of Figure 6), the C₈H₈-fullerene connection involves two adjacent carbon atoms on the double bond of the C₆₀ molecule connecting two hexagons, just as in the case of the high-pressure high-temperature (HPHT) C₆₀ polymers. ¹⁹ The difference is that while in those polymers a cyclobutane ring is formed on polymerization; here, both a four-center (cyclobutane) and a three-center (cyclopropane) ring is possible. The second type of fullerene- C_8H_8 connection (patterns c and e in the third column of Figure 6.) is formed again by two atoms of C₆₀, but these lie on pentagon-hexagon bonds. It has been shown that such a connection pattern can only exist if the ball is opened.²⁷ As an opening was excluded on the basis of IR results, patterns c and e can be eliminated. The last type of connection between a fullerene and a C_8H_8 is a single bond (patterns f, g, and h in the fourth column of Figure 6).

Next, we subject these remaining structures to closer scrutiny. Pattern a was observed in the linear orthorhombic C₆₀ polymer, and b was observed in the two-dimensional tetragonal polymer.¹⁹ In these polymers and in the C₆₀ dimer, an empirical relation holds between the shift of the A_{g} (2) mode and the number of bonds on a single C₆₀ ball: the shift is 5 cm⁻¹ for every cycloaddition connection (i.e., two adjacent single bonds).²⁰ The softening occurs because the bonds formed in the polymerization reaction originate from the π bonds of the fullerene. The shift of 10 cm^{-1} in $poly(C_{60}C_8H_8)$ fits perfectly to pattern a. As the half width of the measured peak is 7 cm^{-1} , it is highly unlikely that pattern b or pristine C_{60} are present in poly- $(C_{60}C_8H_8).$

We can rule out that each fullerene is connected to six cubanes. In this case, because of the stoichiometry, the C₈H₈ molecule should also show sixfold coordination, which would lead to a steric tension with six of the eight C atoms of the hydrocarbon bound to a C_{60} molecule. Therefore, structures d, f, g, and h would automatically imply structure a to be present

According to our knowledge, no fullerene compounds with the connection patterns d, f, g, and h have been thoroughly investigated by vibrational spectroscopy so far. A similar wellknown structure only appears in the case of pattern d: the twodimensional rhombohedral C_{60} polymer¹⁹ has six pairs of σ bonds on hexagon-hexagon bonds of the C₆₀ molecule, although arranged in a different way. The rhombohedral polymer shows the A_g (2) peak at 1406 cm⁻¹ (ref 28). We can expect a shift of similar magnitude in the case of pattern d, but a peak with such a shift was not observed. Another argument which confirms the absence of pattern d comes from the polarization dependence of the Raman spectrum. If poly- $(C_{60}C_8H_8)$ contained only fullerenes with T_h symmetry, then the spectrum should show totally polarized modes, which is not the case. If, on the other hand, it contained fullerenes with different connection patterns and pattern d were one of these, then the peaks should shift or at least change their shape as we change the polarization. As this was not observed either, we can again come to the conclusion that pattern d is not present in poly($C_{60}C_8H_8$).

Up to this point, we derived that $poly(C_{60}C_8H_8)$ definitely contains fullerene units with connection pattern a, but the possibility of patterns f, g, h, or all three cannot be unambigously excluded. If more connection patterns are present, then many newly activated modes should appear, which would lead to a very rich spectrum, like, for example, that of the C₆₀ photopolymer.²⁹ This is in contradiction to the observed spectra. The presence of sixfold, besides twofold, coordinated C₆₀ would also mean that in the frequency region of the A_g , H_g , and T_{1u} modes we would have to see at least 2, 8, and 5 modes, respectively. Instead, we only see somewhat broader peaks as usual. The only remaining possibility would be that all of the Raman and infrared modes of the sixfold coordinated C₆₀ units behave in a very similar way to those of the units with pattern a, which would lead to unobservable splitting. This is very unlikely since the fullerene-C₈H₈ bonds in the two cases are different. Thus, on the basis of our infrared and Raman measurements, we propose that poly(C₆₀C₈H₈) consists of C₈H₈ molecules and fullerene molecules connected according to pattern a.

The twofold coordination of the fullerene unit means that the C₈H₈ unit also has a coordination number of two leading to a structure consisting of chains. We cannot derive a definite assignment as to the structure of the cubane isomer connecting two fullerenes. One possible product, dihydropentalene, would lead to linear chains, but there are possibilities to introduce a 90° turn as well. The simultaneous appearance of the two would introduce disorder in all directions, leading to the cubic and amorphous crystal structure in accordance with X-ray diffraction. The variety in the connecting cubane isomers would also explain the broadening of the vibrational lines.

We can also relate the above conclusions to the structural data on C₆₀•C₈H₈ polymerized at various temperatures and pressures.⁵ Iwasiewicz-Wabnig et al. found two different polymer structures depending on polymerization temperature and pressure: a pseudo-cubic and a pseudo-orthorhombic one. They concluded from Raman spectroscopy that the two do not differ significantly on the molecular level, but the pseudo-

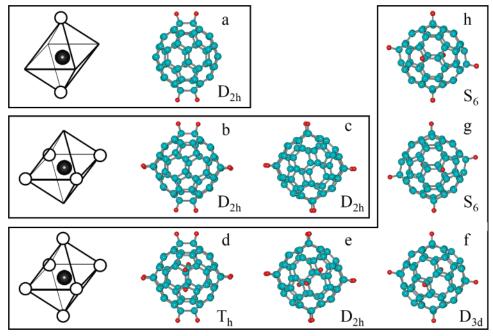


Figure 6. Possible connection patterns of the fullerene in $poly(C_{60}C_8H_8)$. The first column shows the arrangement of C_8H_8 molecules (white spheres) which connect to a C_{60} ball (gray sphere). In the next columns, the carbon atoms of fullerene origin are colored blue, and those of cubane origin are red. We assumed in this scheme that the connection is four-centered, including two atoms of cubane origin. The point group of the fullerene unit is indicated.

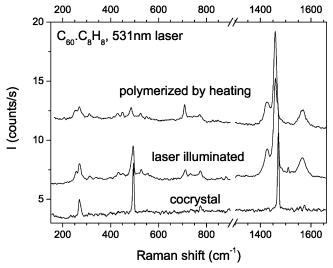


Figure 7. Raman spectrum of $poly(C_{60}C_8H_8)$ after photochemical reaction compared with the spectrum of the cocrystal and the spectrum of the copolymer obtained by annealing.

orthorhombic form is more ordered since its formation occurs at pressures where the rotation of the fullerene balls is sterically hindered. This leads us to believe that the D_{2h} symmetry, compatible with the orthorhombic crystal structure, is intrinsic to the polymer, and the pseudo-cubic allotrope results from a disordered arrangement of these molecular units.

3.3. Photochemical Reaction in C₆₀•C₈H₈. We observed a reaction between the constituents on illumination at room temperature similar to that taking place on heating. After already 100 s of laser illumination in the Raman microscope at both 531 and 468 nm, the intensity of the Raman peak at 1469 cm⁻¹ decreases, and a new line at 1459 cm⁻¹ appears. The Raman spectrum obtained after about an hour of illumination by the 531 nm laser is depicted in Figure 7. The new features in the spectrum coincide with those of the polymer produced by annealing. However, as we will see later, the polymerization

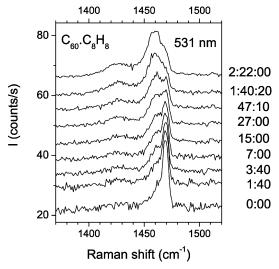


Figure 8. Change of the Raman spectrum of C_{60} $\cdot C_8H_8$ on illumination by the 531 nm laser. The time (in hours:minutes:seconds) of the illumination is indicated on the right-hand side.

here is not triggered by laser-induced heating. Unfortunately, we do not observe any cubane vibrations when exciting with the laser lines at 531 and 468 nm, so we do not know whether cubane isomerizes the same way as in the thermal polymerization process; we can only deduce that the connection pattern of the fullerene is identical.

The gradual evolution of the new spectral pattern around the A_g (2) mode during illumination is illustrated in Figure 8. We fitted the spectra with three Lorentzians: one for the A_g (2) mode of the monomer, one for the A_g (2) mode of the polymer, and one for the H_g (7) mode of the polymer. From the obtained integrated intensity values, the intensity of the polymer A_g (2) peak normalized to the total intensity of the two A_g (2) peaks was calculated. We repeated the procedure for three exciting laser wavelengths: 531, 468, and 676 nm (see Figure 9). We found that longer wavelength laser lines (676 nm or 785 nm)

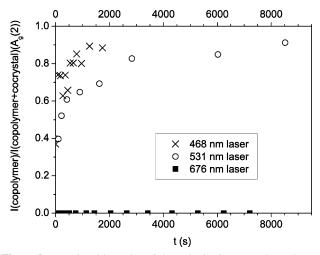


Figure 9. Fractional intensity of the poly($C_{60}C_8H_8$) A_g (2) peak as a function of illumination time for three different lasers.

did not induce the reaction; therefore, the effect of laser heating can be excluded. The wavelength dependence is analogous to that in C_{60} , where photopolymerization takes place on illumination.²² On the basis of these analogies, we classify the reaction as photocopolymerization with excitation of C_{60} as the first step. (We note that the photochemical reaction is also the reason why the accumulation time for the spectrum of the C₆₀•C₈H₈ cocrystal taken at 531 nm (Figure 2) had to be shorter than for that taken at 785 nm (Figure 1), which accounts for the poorer statistics of the former spectrum.)

3.4. Poly($C_{70}C_8H_8$). In $C_{70}\cdot C_8H_8$, a similar irreversible change as in C₆₀•C₈H₈ takes place on heating to 470 K. We show the Raman and IR spectra of the reaction product in Figures 4 and 5 and list the center frequencies of the peaks in Tables 3 and 4, along with the assignments of C_{70} modes by Stratmann et al.¹⁰ The reaction leads to the disappearance of the cubane peaks from both the IR and the Raman spectra, and a new peak appears at 2946 cm⁻¹ in the IR spectrum. At the same time, the IR lines of the fullerene split, but the splitting is much less than that in the C₆₀ analogue. The Raman lines only broaden, probably due to unresolved splitting.

We found that below 800 cm⁻¹ the splitting is twofold in the case of doubly degenerate E'_1 modes. Above 800 cm⁻¹, no clear splitting can be seen, but the lines become somewhat smeared out. From the apparent twofold splitting of the lowfrequency E'_1 modes, the loss of the fivefold axis can be concluded, corresponding to the point group of C_{70} being $C_{2\nu}$ or one of its subgroups.

The changes in the IR spectra of C₇₀•C₈H₈ on annealing reveal a reaction in which the cubane structure changes completely. The resulting hydrocarbon bonds to C₇₀, whose cage distorts, but remains intact. As the reaction product is insoluble in common solvents,¹ it must indeed be a polymer. At this stage of the research, we cannot say anything more about the structure of this polymer, which is partly due to the scarcity of sound spectroscopic results on C₇₀ derivatives and partly due to the more complicated structure of C₇₀.

4. Conclusions

The IR and Raman spectra of C₆₀•C₈H₈ and C₇₀•C₈H₈ were measured both in their rotor-stator and in their polymer phases. The rotor-stator nature of the cocrystals directly manifests itself in the spectra being simple superpositions of those of the constituents. Hence, van der Waals forces are the exclusive interaction between the static cubane and the rotating fullerene

molecules. The slightly lower frequency of the cubane lines can be explained on the basis of the highly polarizable environment of the cubane molecules in these structures.

In the IR and Raman spectra of the polymer phases, the fullerene lines are split and new lines appear, corresponding to a symmetry lowering of the fullerene molecules while their cage remains intact. As the cubane lines change dramatically during the polymerization, we conclude that the cubane isomerizes to another constitutional isomer, which binds to the fullerenes. According to the vibrational spectra, no C₆₀-C₆₀ bonding occurs. The comparison of structural and spectroscopic results allows us to identify linear chains connected via the apical cubane as the most probable polymerization pattern in poly-(C₆₀C₈H₈), with possibly another cubane isomer introducing occasional 90° turns in the chains.

Finally, we found a photochemical reaction in C₆₀•C₈H₈ under illumination with green or blue light. The symmetry of the fullerene molecules in the product turns out to be the same as that in the thermopolymer.

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References and Notes

- (1) Pekker, S.; Kováts, É.; Oszlányi, G.; Bényei, G.; Bortel, G.; Jalsovszky, I.; Jakab, E.; Borondics, F.; Kamarás, K.; Bokor, M.; Kriza, G.; Tompa, K.; Faigel, G. Nat. Mater. 2005, 4, 784.
- (2) Bortel, G.; Faigel, G.; Kováts, É.; Oszlányi, G.; Pekker, S. Phys. Stat. Sol. B 2006, 243, 2999
- (3) Pekker, S.; Kováts, É.; Oszlányi, G.; Bényei, G.; Klupp, G.; Bortel, G.; Jalsovszky, I.; Jakab, E.; Borondics, F.; Kamarás, K.; Faigel, G. Phys. Stat. Sol. B 2006, 243, 3032.
- (4) Kováts, E.; Klupp, G.; Jakab, E.; Pekker, Á.; Kamarás, K.; Jalsovszky, I.; Pekker, S. Phys. Stat. Sol. B 2006, 243, 2985
- (5) Iwasiewicz-Wabnig, A.; Sundqvist, B.; Kováts, É.; Jalsovszky, I.; Pekker, S. Phys. Rev. B 2007, 75, 024114.
 - (6) Eaton, P. E.; Cole, T. W. J. Am. Chem. Soc. 1964, 86, 3157.
- (7) Della, E. W.; McCoy, E. F.; Patney, H. K.; Jones, G. L.; Miller, F. A. J. Am. Chem. Soc. 1979, 101, 7441.
- (8) Bethune, D. S.; Meijer, G.; Tang, W. C.; Rosen, H. J.; Golden, W. G.; Seki, H.; Brown, C. A.; de Vries, M. S. Chem. Phys. Lett. 1991, 179,
 - (9) Sun, G.; Kertesz, M. J. Phys. Chem. A 2002, 106, 6381.
- (10) Stratmann, R. E.; Scuseria, G. E.; Frisch, M. J. J. Raman Spectrosc. 1998, 29, 483.
- (11) Kamarás, K.; Hadjiev, V. G.; Thomsen, C.; Pekker, S.; Fodor-Csorba, K.; Faigel, G.; Tegze, M. Chem. Phys. Lett. 1993, 202, 325.
- (12) Matus, M.; Kuzmany, H.; Krätschmer, W. Solid State Commun. **1991**, 80, 839
- (13) Cole, T. W., Jr.; Perkins, J.; Putnam, S.; Pakes, P. W.; Strauss, H. L. J. Phys. Chem. 1981, 85, 2185.
 - (14) Abe, H.; Yamada, K. M. T. Struct. Chem. 2003, 14, 211.
- (15) Vougioukalakis, G. C.; Prassides, K.; Orfanopoulos, M. Org. Lett. 2004, 6, 1245.
- (16) Hassenrück, K.; Martin, H.-D.; Walsh, R. Chem. Rev. 1989, 89, 1125.
- (17) Ishida, H.; Komori, K.; Itoh, K.; Ohno, M. Tetrahedron Lett. 2000, 41, 9839.
- (18) Zhou, P.; Dong, Z.-H.; Rao, A. M.; Eklund, P. C. Chem. Phys. Lett. 1993, 211, 337.
- (19) Núnez-Regueiro, M.; Marques, L.; Hodeau, J.-L.; Bethoux, O.; Perroux, M. Phys. Rev. Lett. 1995, 74, 278.
- (20) Wågberg, T.; Jacobsson, T.; Sundqvist, B. Phys. Rev. B 1999, 60,
- (21) Oszlányi, G.; Bortel, G.; Faigel, G.; Gránásy, L.; Bendele, G. M.; Stephens, P. W.; Forro, L. Phys. Rev. B 1996, 54, 11849.
- (22) Rao, A. M.; Zhou, P.; Wang, K.; Hager, G. T.; Holden, J. M.; Wang, Y.; Lee, W. T.; Bi, X.; Eklund, P. C.; Cornett, D. S.; Duncan, M. A.; Amster, I. J. Science 1993, 259, 955.

- (23) Lebedkin, S.; Gromov, A.; Giesa, S.; Gleiter, R.; Renker, B.; Rietschel, H.; Krätschmer, W. *Chem. Phys. Lett.* **1998**, 285, 210.
- (24) Krause, M.; Dunsch, L.; Seifert, G.; Fowler, P. W.; Gromov, A.; Krätschmer, W.; Gutierez, R.; Porezag, D.; Frauenheim, T. J. Chem. Soc., Faraday Trans. 1998, 94, 2287.
- (25) Klupp, G.; Borondics, F.; Oszlányi, G.; Kamarás, K. AIP Conf. Proc. 2003, 685, 62.
- (26) Long, V. C.; Schundler, E. C.; Adams, G. B.; Page, J. B.; Bietsch, W.; Bauer, I. Phys. Rev. B 2007, 75, 125402.
- (27) Schick, G.; Hirsch, A. *Tetrahedron* 1998, 54, 4283.
 (28) Davydov, V. A.; Kashevarova, L. S.; Rakhmanina, A. V.; Senyavin, V. M.; Ceolin, R.; Szwarc, H.; Allouchi, H.; Agafonov, V. Phys. Rev. B **2000**, *61*, 11936.
 - (29) Onoe, J.; Takeuchi, K. Phys. Rev. B 1996, 54, 6167.