See discussions, stats, and author profiles for this publication at: https://www.researchgate.net/publication/23423809

Adsorption properties and structure of CO(2) adsorbed on open coordination sites of metalorganic framework Ni(2)(dhtp) from gas adsorption, IR spectroscopy and X-ray diffraction

ARTICLE in CHEMICAL COMMUNICATIONS · DECEMBER 2008

Impact Factor: 6.83 · DOI: 10.1039/b810574j · Source: PubMed

CITATIONS

176

READS

135

7 AUTHORS, INCLUDING:



Rune E. Johnsen

Technical University of Denmark

25 PUBLICATIONS 486 CITATIONS

SEE PROFILE



Helmer Fjellvåg

University of Oslo

227 PUBLICATIONS 4,288 CITATIONS

SEE PROFILE



Sachin Maruti Chavan

University of Oslo

38 PUBLICATIONS 1,791 CITATIONS

SEE PROFILE



Richard Blom

SINTEF, Oslo, Norway

102 PUBLICATIONS 3,422 CITATIONS

SEE PROFILE

Adsorption properties and structure of CO₂ adsorbed on open coordination sites of metal-organic framework Ni₂(dhtp) from gas adsorption, IR spectroscopy and X-ray diffraction[†]

Pascal D. C. Dietzel,^a Rune E. Johnsen,^b Helmer Fjellvåg,^b Silvia Bordiga,^c Elena Groppo, Sachin Chavan and Richard Blom*

Received (in Cambridge, UK) 23rd June 2008, Accepted 9th September 2008 First published as an Advance Article on the web 24th September 2008 DOI: 10.1039/b810574j

The microporous metal-organic framework Ni₂(dhtp) (H₄dhtp 2,5-dihydroxyterephthalic acid) shows distinct end-on CO₂ coordination to coordinatively unsaturated nickel sites giving rise to high CO2 adsorption capacity at sub-atmospheric pressures and ambient temperatures.

Separation of CO₂ from flue gases is challenging due to the relatively low concentrations of CO₂, typically from 4 to 12 volume% depending on the fuel used in the process. At present, the most developed technology for flue gas separation is absorption of CO₂ into solvents, often based on aqueous alkanolamines, such processes having high energy demand and potential unwanted environmental consequences due to solvent loss and degradation. A step towards dry processes utilizing either membranes or adsorbents might give both environmental and energetic benefits. However, in the development of adsorption based processes one major challenge is to develop adsorbents having high adsorption capacity for CO₂ at the actual low partial pressures. Most promising low pressure adsorbents reported so far are based on the deposition of nitrogen containing oligomeric or polymeric species onto high surface area silicas,2 having initial adsorption capacities up to 18 wt% CO₂ at around 0.1 atm CO₂ pressure and 298 K. However, for some of these the capacities drop to below 10 wt% after the first adsorption cycle.³ For metalorganic framework based adsorbents most focus has been put on the high pressure adsorption capacity of CO₂,⁴ and capacities up to 150 wt% CO2 at 40 bar and ambient temperature have been reported.⁵ The low pressure adsorption capacity of CO2 on MOFs is in most cases below 5 wt% and for some MOFs a high external gas pressure is needed to open up the flexible MOF structure for further adsorption. One important advantage with MOFs as compared to most polymer based adsorbents is that MOFs form crystalline solids. Thus,

In the present communication we report the structure of the adsorbate on the single-site adsorption site in Ni₂(dhtp),⁷ based on X-ray diffraction data and IR spectroscopy. Adsorption experiments show that this MOF has an adsorption capacity of CO₂ of 11 wt% even at 0.1 atm CO₂ pressure and 303 K which is among the highest CO₂ adsorption capacity reported for MOFs at these conditions.

[Ni₂(dhtp)(H₂O)₂]·8H₂O (CPO-27-Ni) was prepared as described previously.⁷ Prior to the adsorption experiments the powder was activated by thermal treatment in a vacuum at 383 K overnight. From nitrogen adsorption experiments at 77 K the Langmuir and BET surface areas were determined to be 1312 and 1083 m² g⁻¹, respectively. The same sample was used to measure the CO₂ isotherm at temperatures from 303 to 353 K. Fig. 1 shows the isotherms as well as the deduced differential enthalpy of adsorption using the isosteric method from vacuum to a CO₂ pressure of 1 atm. The steepness of the slope in the low pressure part of the isotherms collected at 303 and 313 K indicate strong adsorption, while the isotherm at 353 K is noticeably less steep in this region. At low pressures the differential enthalpy of adsorption is estimated to be around 40 kJ mol⁻¹ up to around p > 0.2 atm. It subsequently decreases slightly at higher pressures. Theoretically, full coverage of CO2, i.e. one CO2 bonded to each open nickel coordination site, corresponds to 28.2 wt% CO₂. Consequently, a $\sim 80\%$ coverage is reached at a CO₂ pressure of 1 atm in the isotherm collected at 303 K. Even higher degrees of coverage have recently been obtained in the case of NO,8 in which a

[†] Electronic supplementary information (ESI) available: Details of the structure refinement. CCDC reference number 692602. For ESI and crystallographic data in CIF or other electronic format see DOI: 10.1039/b810574j

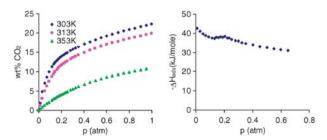


Fig. 1 Adsorption isotherms (left) and differential adsorption enthalpies (right) for the adsorption of CO2 on CPO-27-Ni at the temperatures indicated.

structural information about the mechanism of adsorption is possible through XRD analyses. From such studies the structure of the adsorption site(s) can be obtained which subsequently can be used for further adsorbent optimization.

^a SINTEF Materials & Chemistry, P.O. Box 124 Blindern, N-0314 Oslo, Norway. E-mail: Richard.blom@sintef.no; Fax: +47.22067350

^b Department of Chemistry, University of Oslo, P.O. Box 1033 Blindern, N-0315 Oslo, Norway

^c Dipartimento di Chimica IFM and NIS Centre of Excellence, Via P. Giuria 7, I-10125 Torino, Italy

strong coordinative bond is found between the NO and the metal cation of the MOF, as indicated by a crystal structure determination of the NO adduct of CPO-27-Co. The difference in adsorption capacities between the isotherms collected at 303 and 353 K indicates a possible temperature swing process cycling between these two temperatures having an effective CO_2 capacity of around 10 wt% for p > 0.1 atm.

In order to study the structure of the adsorbate–adsorbent system, a powder sample was prepared in a capillary and flame sealed at a partial pressure of 0.2–0.5 atm, corresponding to 50–65% coverage at room temperature, and measured at 100 K at the Swiss-Norwegian Beamlines at the ESRF using a high resolution powder X-ray diffractometer. The crystal structure of the host–guest compound was then determined by Rietveld refinement.‡

The structure of the dehydrated framework $Ni_2(dhtp)$ was used as a starting model. The atoms of the CO_2 molecule were subsequently localised from difference Fourier maps (full details are given in the ESI†). The structure of the three-dimensional network of CPO-27-M (M = Co, Ni, Zn, Mg), 7,10,11 which is isostructural to MOF-74, 12 resembles a honeycomb. The hydrated compound contains metal atoms which are octahedrally surrounded by oxygen atoms of the organic ligands and a solvent molecule. After removal of the solvent molecule, the metal atoms have a square-pyramidal environment formed by five oxygen atoms of the organic linking molecules and a sixth coordination site available for coordination of adsorbates. 7,10,13

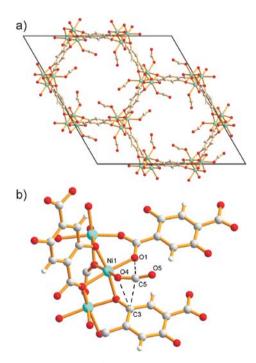


Fig. 2 Crystal structure of CPO-27-Ni with adsorbed CO_2 at the metal atom. (a) Packing along [001] showing the end-on coordination of the CO_2 molecule; (b) local environment of the adsorbed CO_2 molecule. The black dashed lines indicate selected distances longer than covalent bonds and shorter than 3.2 Å (corresponding to the sum of the van der Waals radii of oxygen and carbon) which might influence the structure of the CO_2 molecule through crystal packing effects or back-donation.

In the structure of the CPO-27-Ni/CO₂ complex (Fig. 2), the CO₂ molecule is clearly coordinating the Ni²⁺ atom in the framework in end-on fashion with a Ni–O distance of 2.29(2) Å, a Ni–C distance of 3.04(2) Å, and a Ni–O–C angle of 117(2)°. The Ni–O distance is significantly shorter than the sum of the van der Waals radii of Ni and O of 3.1 Å, and it therefore attests to the strength of interaction between the metal adsorption site of the framework and carbon dioxide. The carbon dioxide molecule appears to be slightly distorted from its usual linear structure with an O–C–O angle of 162(3)°. However, notice that the estimated standard deviations given by Rietveld refinements are underestimated and should be multiplied with a factor of at least 3 to reach a realistic value.¹⁴

IR spectroscopy is a complementary method which can be used to study the interaction of carbon dioxide with the Lewis acid sites of such material. CO_2 can interact with the cations through one of the oxygen lone pair orbitals with formation of end-on adducts, $^{15-17}$ though it is generally weakly bonded at room temperature.

Fig. 3 shows the background subtracted IR spectra obtained by progressive lowering of equilibrium pressure at room temperature, the highest coverage spectrum (grey curve) corresponding to an equilibrium pressure of 0.02 atm. Starting the discussion from low coverage, we observe the growth of the main component at 2341 cm⁻¹ with a low frequency shoulder at 2329 cm⁻¹. Both bands quickly go off the scale without reaching saturation. This band can be readily assigned to the ν_3 mode of carbon dioxide forming Ni²⁺···O=C=O adducts with end-on configuration. The bands due to the ν_3 mode of CO_2 are slightly red shifted by $\Delta \nu = -8 \text{ cm}^{-1}$ with respect to gas phase (2349 cm⁻¹) as previously observed in case of HKUST-1. On the low frequency side of this main absorption we observe the growth of less intense bands, centred around 2275 cm⁻¹, due to Ni²⁺···O=¹³C=O which is present naturally in ¹²CO₂ (1%). The broad and weak band centred at 2408 cm⁻¹ can be associated with the combination mode $\nu_3 + \nu_{M-O}$, where ν_{M-O} denotes the Ni²⁺···O stretching of the Ni²⁺···O=C=O adduct. The ν_{M-O} mode would be expected at about 70 cm⁻¹. ¹⁸In the low frequency range, an intense band is expected for doubly degenerate bending mode (ν_2) of CO₂ (gas frequency at 668 cm⁻¹). In this case (see the

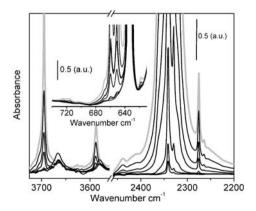


Fig. 3 FTIR spectra (background subtracted) of CO₂ adsorbed on CPO-27-Ni at decreasing equilibrium pressure (higher coverage, grey curve corresponds to 0.02 atm equilibrium pressure). In the inset the CO₂ bending region is reported.

inset in Fig. 3), a doublet at 659 and 651 cm⁻¹ is clearly visible, testifying that the molecule is weakly perturbed and also might be slightly distorted.

In the high frequency region (3800–3500 cm⁻¹), two components are observed at 3666 and 3580 cm⁻¹. The nature of these two absorptions is well known and is interpreted on the basis of the combination of ν_1 and ν_3 modes of CO₂. The reason for the presence of a doublet shifted by about $\pm 50 \text{ cm}^{-1}$ from the expected frequency for the combination mode ν_1 and ν_3 , is due to the fact that the first overtone of the ν_2 mode coincides with the ν_1 mode, causing a strong Fermi resonance effect that induces a band splitting in two components $100~{\rm cm}^{-1}$ apart. 19,20

In contrast to what is observed in the case of metal complexes, where experimental structural evidence for CO₂ coordination via $\eta^1(C)$ (bonded through the C atom) and η²(C,O) modes has been previously reported, ^{21,22} coordination of CO₂ via η^{1} (O) to metal ions is reported here for the first time. The spectroscopic data can be explained only with an end-on coordination of CO₂, and a slight bending of the CO₂ molecule as suggested by XRD can be hypothesized. A bending can be explained by considering the steric and electronic environment around the coordinated carbon dioxide: in particular the short C5···O1 interaction of 3.00(3) Å may indicate a secondary interaction between the framework O1 atom and the C atom of CO₂ leading to a slight bending in a similar manner as previously suggested to take place in zeolites²³ and in HKUST-1.19 In the other metal-organic framework for which crystal structural information of the CO₂ coordination is available, MIL-53(Cr), the carbon dioxide interacts with an OH group of the inorganic building block.²⁴

In conclusion, we have demonstrated the high CO₂ uptake of coordination polymer CPO-27-Ni at ambient and elevated temperatures, which may make this material applicable in carbon dioxide removal processes. Our adsorption results are consistent with a very recent publication by Matzger et al. on the CO₂ adsorption on a series of isostructural M₂(dhtp) materials,²⁵ which has appeared in press after submission of this manuscript. We have in addition shown that the strong adsorption is caused by a direct interaction between CO₂ and the open Lewis acid site on nickel where CO₂ is end-on coordinated to the nickel cation of the framework. The strong interaction is exemplified by the short Ni-O distance in the adduct. Both, powder X-ray diffraction and IR-spectroscopy indicate that the CO₂ molecule might be distorted slightly from linear shape.

We gratefully acknowledge Wouter van Beek and Hermann Emmerich for assistence provided at the Swiss-Norwegian Beamlines, ESRF. This work was supported by EU FP6 NMP STREP "MOFCAT" (033335) and by the Research Council of Norway (NANOMAT contract 182056/S10).

Notes and references

‡ The powder sample of the CPO-27-Ni/CO2 was measured at the Swiss-Norwegian Beamlines (BM01B) at the European Synchrotron Radiation Facility (ESRF) in Grenoble, France. Rietveld refinement was performed with the WINPOW program. Details of the refinement process are given in the ESI.† $C_4H_1NiO_3\cdot(CO_2)_{0.67}$, M=185.4 g mol⁻¹, trigonal space group $R\overline{3}$, a=b=25.7836(11) Å, c = 6.7474(3) Å, $\alpha = \beta = 90^\circ$, $\gamma = 120^\circ$, V = 3884.7(5) Å³, Z = 18, $\lambda = 0.520054$ Å, T = 100 K, 2θ range $1.027-25.499^\circ$, step size 0.004° , $R_{\rm p} = 0.0590, \ R_{\rm wp} = 0.0814, \ R_{\rm exp} = 0.0264, \ R_{\rm Bragg} = 0.0155,$ GoF = 3.09.

- 1 A. B. Rao and E. S. Rubin, Environ. Sci. Technol., 2002, 36, 4467.
- 2 TEPAN (tetraethylenepentamine) bases adsorbents: T. Filburn, J. J. Helble and R. A. Weiss, Ind. Eng. Chem. Res., 2005, 44, 1542; PEI (polyethyleneimine) based adsorbents: X. Xu, C. Song, J. M. Andresen, B. G. Miller and A. W. Scaroni, Microporous Mesoporous Mater., 2003, 62, 29.
- 3 M. L. Gray, Y. Soong, K. J. Champagne, H. Pennline, J. P. Baltrus, R. W. Stevens, Jr, R. Khatri, S. S. C. Chuang and T. Filburn, Fuel Proc. Technol., 2005, 86, 1449.
- 4 K. Uemura, R. Matsuda and S. Kitagawa, J. Solid State Chem., 2005, 178, 2420; S. Bourrelly, P. L. Liewellyn, C. Serre, F. Millange, T. Loiseau and G. Férey, J. Am. Chem. Soc., 2005, **127**, 13519.
- 5 A. R. Millward and O. M. Yaghi, J. Am. Chem. Soc., 2005, 127, 17998.
- 6 A. J. Fletcher, K. M. Thomas and M. J. Rosseinsky, J. Solid State Chem., 2005, 178, 2491; N. A. Ramsahye, G. Maurin, S. Bourrelly, P. L. Llewellyn, T. Loiseau, C. Serré and G. Férey, Chem. Commun., 2007, 3261; K. Uemura, Y. Yamasaki, Y. Komagawa, K. Tanaka and H. Kita, Angew. Chem., Int. Ed., 2007, 46, 6662
- 7 P. D. C. Dietzel, B. Panella, M. Hirscher, R. Blom and H. Fjellvåg, Chem. Commun., 2006, 959.
- F. Bonino, S. Chavan, J. G. Vitillo, E. Groppo, G. Agostini, C. Lamberti, P. D. C. Dietzel, C. Prestipino and S. Bordiga, Chem. Mater., 2008, 20, 4957
- A. C. McKinlay, B. Xiao, D. S. Wragg, P. S. Wheatley, I. L. Megson and R. E. Morris, J. Am. Chem. Soc., 2008, 130, 10440-10444.
- 10 P. D. C. Dietzel, Y. Morita, R. Blom and H. Fjellvåg, Angew. Chem., Int. Ed., 2005, 44, 6358; P. D. C. Dietzel, R. E. Johnsen, R. Blom and H. Fjellvåg, Chem.-Eur. J., 2008, 14, 2389.
- 11 P. D. C. Dietzel, R. Blom and H. Fjellvåg, Eur. J. Inorg. Chem., 2008 3624
- 12 N. L. Rosi, J. Kim, M. Eddaoudi, B. Chen, M. O' Keeffe and O. M. Yaghi, J. Am. Chem. Soc., 2005, 127, 1504.
- 13 J. G. Vitillo, L. Regli, S. Chavan, G. Ricchiardi, G. Spoto, P. D. C. Dietzel, S. Bordiga and A. Zecchina, J. Am. Chem. Soc., 2008, 130, 8386
- 14 R. J. Hill and L. M. D. Cranswick, J. Appl. Crystallogr., 1994, 27, 802.
- 15 G. Busca and V. Lorenzelli, Mater. Chem., 1982, 7, 89
- 16 C. Morterra, G. Cerrato and C. Emanuel, Mater. Chem. Phys., 1991, 29, 425; B. Bonelli, B. Civalleri, B. Fubini, P. Ugliengo, C. Otero Area'n and E. Garrone, J. Phys. Chem., 2000, 104, 10978.
- 17 B. Bonelli, B. Onida, B. Fubini, C. Otero Areán and E. Garrone, Langmuir, 2000, 16, 4976.
- 18 F. X. Llabrés i Xamena and A. Zecchina, Phys. Chem. Chem. Phys., 2002, 4, 1978
- 19 S. Bordiga, L. Regli, F. Bonino, E. Groppo, C. Lamberti, B. Xiao, P. S. Wheatley, R. E. Morris and A. Zecchina, Phys. Chem. Chem. Phys., 2007, 9, 2676.
- 20 S. Usseglio, A. Damin, D. Scarano, S. Bordiga, A. Zecchina and C. Lamberti, J. Am. Chem. Soc., 2007, 129, 2822.
- 21 D. H. Gibson, Coord. Chem. Rev., 1999, 185-186, 335, and references therein.
- J. Mascetti, F. Galan and I. Pàpai, Coord. Chem. Rev., 1999, 190-192, 557
- 23 E. Garrone, B. Bonelli, C. Lamberti, B. Civalleri, M. Rocchia, P. Roy and C. Otero Areán, J. Chem. Phys., 2002, 117, 10274.
- 24 C. Serre, S. Bourrelly, A. Vimont, N. A. Ramsahye, G. Maurin, P. L. Llewellyn, M. Daturi, Y. Filinchuk, O. Leynaud, P. Barnes and G. Férey, Adv. Mater., 2007, 19, 2246.
- 25 S. R. Caskey, A. G. Wong-Foy and A. J. Matzger, J. Am. Chem. Soc., 2008, 130, 10870-10871.