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Well-oriented epitaxial gold nanotriangles and bowties on MoS₂ for surface-enhanced Raman scattering†

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We report on epitaxial growth of Au nanotriangles (AuNTs) and bowties consisting of opposing tip-to-tip AuNTs with a few-nanometer gap on *n*-layer MoS₂. These AuNTs exhibit well-defined crystallographic orientations with the average size determined by the MoS₂ thickness. Monolayer (1L) MoS₂ shows the weakest coalescence, corroborating the layer-dependent interactions among Au and *n*-layer MoS₂. High-resolution transmission electron microscopy characterization confirms the lattice directing of thin MoS₂ sheets on the AuNT formation with [111]_{Au}||[001]_{MoS₂} and <211>_{Au}||<210>_{MoS₂}. By introducing misfit dislocation arrays, the system with an 8.8% lattice misfit is stress-free. In particular, subwavelength-sized gold bowtie nanoantennas can be easily found on the MoS₂ surface with a spacing gap down to 3 nm and a density up to $1.6 \times 10^{13} \text{ m}^{-2}$. This technique is low cost, time-saving and free of impurities compared to the conventional lithography technologies. Meanwhile, the equilateral AuNTs can enhance Raman signals of thin MoS₂ sheets far stronger than that by ordinary gold films, indicating the potential use of AuNTs as SERS substrates for SERS applications.

undergo remarkable changes depending on the number of layers, which are obviously different from its bulk counterpart. For example, with reduced number of layers, size effects have been observed on surface diffusion of metal adatoms on graphene.^{5–7} For *n*-layer MoS₂, it has a tunable bandgap as the number of layers (thickness) decreases, and monolayer MoS₂ becomes a direct band-gap semiconductor in contrast to few-layer or bulk MoS₂ with indirect bandgaps.⁸ In particular, a single layer of MoS₂ can be divided into three weakly bonded S–Mo–S layers with one S layer at the surface through van der Waals force, which is suitable for metal adsorption and binding.^{9,10} Furthermore, the growth and anisotropic etching of MoS₂ single crystals usually exhibit triangular shapes.^{11–13} Thus, due to its single crystalline nature, it is intuitive to imagine that two dimensional layered MoS₂ can be used as templates for the growth of single crystalline epilayers or crystallographically oriented nanostructures. Although significant efforts have been devoted to investigating the electronic and optical properties of layered MoS₂ and its synthesis,^{8,11,14,15} evaluating layered MoS₂'s potential usefulness for epitaxial growth is still in its infancy.¹⁶

Nowadays, the controlled synthesis of AuNTs is of particular interest because of their potential use in cancer hyperthermia, plasmonic nanoantennas,¹⁷ surface enhanced Raman scattering (SERS),¹⁸ etc. Usually, it is essential to assemble them on planar surfaces for many sensing and catalytic applications, which is achieved by post treatment after solution-phase synthesis.^{19,20} In this sense, it is highly desirable to realize the simultaneous growth and assembly of AuNTs without introducing any reagents in the experiments. On the other hand, sub-wavelength-sized gold bowtie nanoantennas, consisting of two opposing tip-to-tip AuNTs separated by a small gap, have attracted great attention due to their applications in SERS or fluorescence,²¹ optical imaging and storage, molecular sensing,²² etc. It is suggested that an extremely high electromagnetic field enhancement can be achieved within the gap and confined to the area near the gap due to the coupled plasmon resonance between two AuNTs, which would serve as

Introduction

Since the discovery of graphene in 2004,¹ considerable efforts have been devoted to two-dimensional layered materials.^{2–4} With the thickness of a layered material reduced to nanoscale, the electronic, magnetic, optical and chemical properties may

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electromagnetically enhanced “hot spots” for SERS applications. The enhancement factors depend on the spacing gap:²³ the smaller the gap is, the stronger the enhancement is. However, there is no direct synthesis approach, but only using expensive e-beam lithography (EBL) to pattern gold bowties, with the spacing gap further limited by the system spatial resolution (~ 8 nm).²³ The advent of two-dimensional layered MoS₂ may fill this gap, which would be an alternative template for growing AuNTs. In this paper, we demonstrated the direct growth and favorable orientation of AuNTs on MoS₂ layers, and aimed to unveil the effect of the number of MoS₂ layers on the AuNT size and areal density. It is interesting to point out that upon heating the substrate, the as-deposited gold adatoms on MoS₂ tend to coalescent into ultrathin equilateral AuNTs, with the particle size and density tuned by the MoS₂ thickness. Few-layer MoS₂ shows better modulation on particle shape than 1L MoS₂, and AuNTs are surprisingly well-oriented on the MoS₂ surface, resulting in the formation of gold bowtie structures with a high density up to 1.6×10^{13} counts per m² and interparticle spacing gap down to 3 nm. The AuNTs grow on MoS₂ with $[111]_{\text{Au}}||[001]_{\text{MoS}_2}$ and $\langle 211 \rangle_{\text{Au}}||\langle 210 \rangle_{\text{MoS}_2}$, suggesting that the orientation of AuNTs originates from the lattice epitaxy between MoS₂ and AuNTs. These results indicate that although the MoS₂ interlayer interaction is usually thought to be weak and short ranged, it has a profound effect on modulating the nucleation and growth of AuNTs on the MoS₂ basal plane. Meanwhile, strong Raman enhancements of AuNTs on MoS₂ are observed, which are much stronger than those based on ordinary gold films, indicating the potential use in SERS.

Results and discussion

Thickness determination of pristine *n*-layer MoS₂

Ultrathin MoS₂ sheets were obtained by mechanical cleavage of bulk MoS₂, and transferred onto 300 nm SiO₂/Si substrates.^{5,12} The samples were first selected based on their color contrasts under an optical microscope. Fig. 1a and b show typical optical images of the as-prepared monolayer (1L), bilayer (2L), trilayer (3L), quadralayer (4L) and multilayer (ML, $n \geq 5$) MoS₂, from which we can see the different color contrasts among MoS₂ flakes with different thicknesses. After selecting thin MoS₂ sheets, we have applied Raman and PL spectroscopy to further determine the number of MoS₂ layers (Fig. 1c and d). It is noteworthy that the frequency difference between two Raman A_{1g} and E_{2g} modes (Fig. 1c) is distinctively different for different number of layers, which increases with the increase of MoS₂ layers. While in the PL spectra (Fig. 1d), the peak intensity of 1L at around 1.84 eV is much more prominent than other *n*-layer MoS₂, which is due to the transition from indirect (few-layer) to direct (1L) band-gap semiconductor. The differences in Raman and PL spectra as a function of thickness are very consistent with previously reported results.^{15,24}

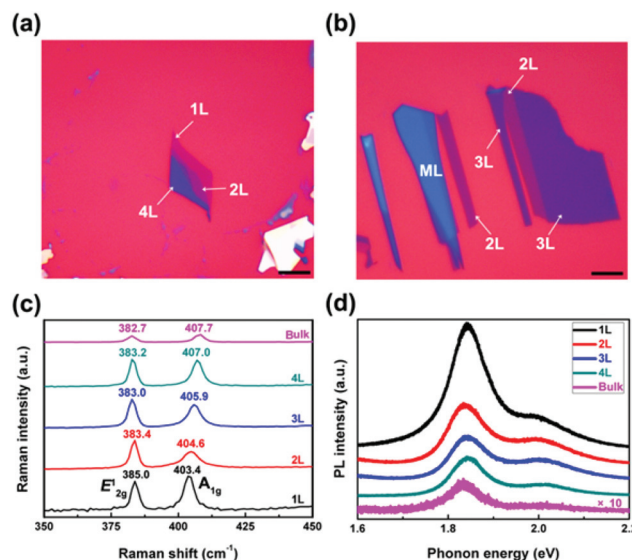


Fig. 1 (a) A representative optical image of 1L, 2L and 4L MoS₂ transferred onto the silicon substrate. (b) An optical image showing the different color contrasts of 2L, 3L and ML MoS₂ on the substrate. Scale bar: 5 μm . (c) and (d) Raman and PL spectra of *n*-layer MoS₂ ($n = 1, 2, 3, 4$ and bulk) excited by 514.5 nm, respectively.

Epitaxial growth of AuNTs or gold bowties on MoS₂

Although ultrathin MoS₂ sheets show hexagonal layered-lattice crystal structures, 1L MoS₂ is truly different from graphene in that it is composed of three atomic layers: a Mo layer sandwiched between two S layers. These would result in special properties on the few-layer MoS₂ surface. To further demonstrate this point, we have thermally evaporated ultrathin gold films onto the pre-heated MoS₂ surface (ESI†), and investigated the effects of MoS₂ thickness on gold morphologies in detail by using scanning electron microscopy (SEM).

By thermally depositing a thin gold film onto the pre-heated MoS₂ surface, we find that the morphology of gold on MoS₂ is layer dependent (Fig. 2). On the one hand, it is surprising to find that gold adatoms tend to nucleate and grow into AuNTs as shown in Fig. 2a and b, with the particle size modulated by the MoS₂ thickness: the thicker the MoS₂ sheet is, the larger the size of AuNTs is (Fig. S1†). In contrast, no AuNTs can be found on bare SiO₂/Si substrates (Fig. 2a). These triangular structures are consistent with recently published results on the growth of MoS₂ single crystals by the chemical vapor deposition (CVD) method,^{12,25} in which single grains tend to grow in triangular shape. The observed triangular shape (Fig. 2) may imply that a special direction of AuNTs is more stable when they are grown on the MoS₂ surface, so as to ensure the smallest surface free energy during growth. Once we deposit a thicker gold film like 4.0 nm, some of the nanoparticles become irregular, and so the effects of MoS₂ thickness on the shape of gold nanostructures become much more prominent (Fig. 2c–e): the AuNPs on 1L MoS₂ are mainly composed of irregular islands, while on thicker MoS₂, like 2L, 3L

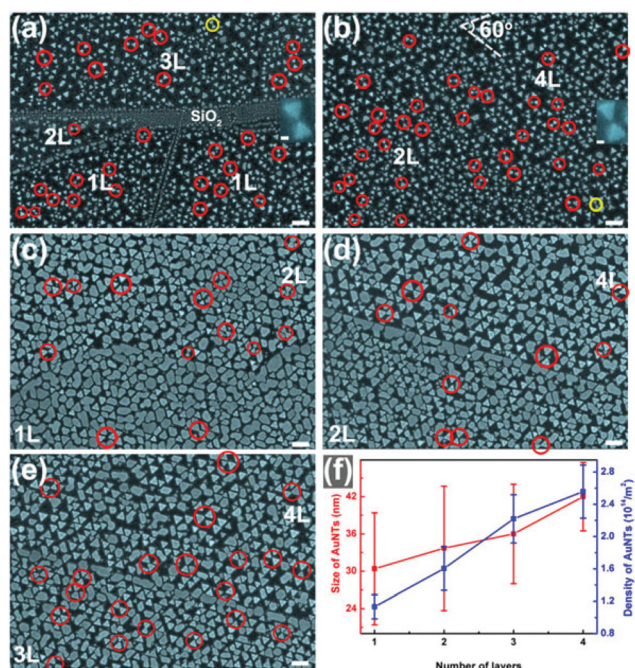


Fig. 2 Typical SEM images showing the growth of ultrathin equilateral AuNTs on *n*-layer MoS₂ with controlled gold film thickness and substrate temperature. (a and b) 2.0 nm at 375 °C. The circles indicate the formation of gold bowties on MoS₂, and the insets show the enlarged sizes of the yellow-marked bowties with 10 nm bar and false color. (c–e) 4.0 nm at 375 °C. (f) The statistical diagram showing the edge length and density of AuNTs as a function of MoS₂ layers as shown in (c)–(e). Scale bar for (a)–(e): 100 nm.

or 4L, more AuNPs exhibit equilateral triangular shape, indicating that 1L MoS₂ is not so suitable for nanotriangle formation compared with other few-layer MoS₂ under the same conditions. In particular, with the increase of MoS₂ thickness, the density of AuNTs becomes much higher, and the size gradually increases (Fig. 2c–e), making it evident that AuNTs are preferentially grown on few-layer rather than on 1L MoS₂, and the growth of triangular AuNPs is truly related to the MoS₂ thickness. To account for the thickness dependence of gold morphologies on the number of MoS₂ layers, we need to consider the nucleation and growth of gold adatoms on the MoS₂ surface, which is determined by the thickness-dependent surface diffusion coefficient and barrier caused by the interlayer interactions as in the situation of gold on the graphene surface.^{5,7} Meanwhile, it is apparent from the images (Fig. 2a–e) that the deposited Au atoms have the tendency to aggregate at MoS₂ edges or at terraced edges where MoS₂ flakes stack together, which is probably attributed to the reason that Au atoms have larger binding energy with the edge Mo/S atoms than that at the central regions. For instance, the AuNPs are nearly connected with each other at the terraced edges between 2L and 4L (Fig. 2d), or 3L and 4L MoS₂ (Fig. 2e), indicating the aggregation of gold atoms at these border lines.

To the best of our knowledge, it is probably the first time gold bowties are directly synthesized on MoS₂ templates without using the lithography method. As shown in Fig. 2a and b, it is worth mentioning that lots of bowtie structures (marked with red circles in Fig. 2a and b) consist of two nano-triangles with tips facing each other for gaps as small as 3 nm (the insets of Fig. 2a and b), which is much smaller than the reported value, and the relevant density can be up to 1.6×10^{13} counts per m². Meanwhile, the gap in the gold bowties is in the range of 3–10 nm, and most of the gaps are around 4–6 nm (Fig. S2†). In addition, the size of gold bowties is closely related to the gold film thickness, and can be tuned through changing the film thickness (Fig. 2). With the increase of film thickness from 2 nm to 4 nm, the edge lengths of AuNTs range from ~20 nm to 40 nm, resulting in the formation of gold bowties with larger spatial size. Finally, we find that the areal density of gold bowties increases with the increase of the density of AuNTs.

Lattice epitaxy between MoS₂ and AuNTs

To gain further insights into the effects of MoS₂ on the nucleation and growth of equilateral AuNTs, we carried out HRTEM observation to characterize the structural relationship between gold and MoS₂ (Fig. 3). The oriented nanotriangles are epitaxially grown on MoS₂. Au has a face-centered cubic (fcc) crystal structure, which is six-fold-symmetric along the [111]-axis,

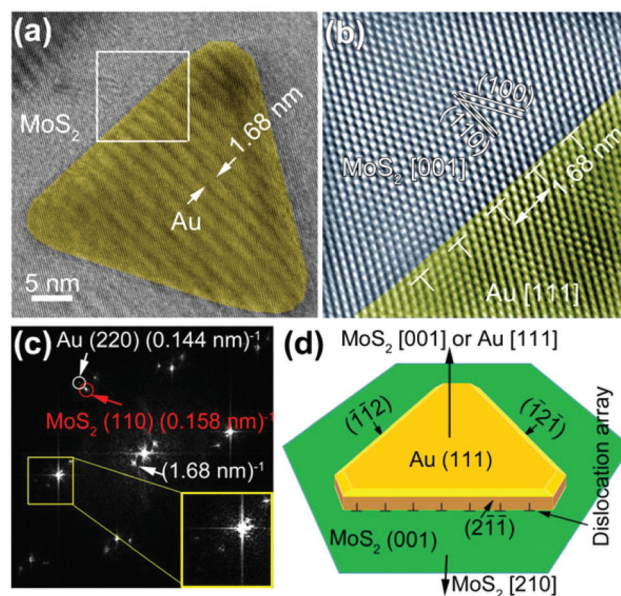


Fig. 3 Epitaxial growth of Au nanotriangles. (a) HRTEM image of a AuNT on MoS₂. (b) Fourier filtered image of the white square in panel (a), showing a dislocation array with a spacing of 1.68 nm at the boundary of the Au triangle. (c) Fourier transformation pattern of the Au triangle, indicating that Au triangles are grown on MoS₂ with $\{220\}_{\text{Au}} \parallel \{110\}_{\text{MoS}_2}$. The diffraction spots in the yellow square exhibit a six-fold symmetry of the dislocations. (d) Schematic of the structural relationship between Au and MoS₂. Note that there are three dislocation arrays that are perpendicular to the $(2\bar{1}1)$, $(\bar{1}2\bar{1})$ and $(\bar{1}\bar{1}2)$ facets of the AuNT, respectively.

while MoS₂ has a hexagonal structure and is also six-fold-symmetric along the [001]-axis. Although the lattice mismatch

$$f = \frac{d_{\text{Au}\{220\}} - d_{\text{MoS}_2\{110\}}}{d_{\text{MoS}_2\{110\}}} \quad (1)$$

between these two materials is large (8.8% between Au {220} and MoS₂ {110}, with a *d*-spacing of 0.144 nm and 0.158 nm, respectively), the AuNTs can grow on the MoS₂ surface by introducing dislocation arrays with a pitch of ~1.68 nm to form stress-free interfaces. That is, dislocations appear every 11.6 Au {220} spacings or every 10.6 MoS₂ {110} spacings. Similar epitaxial growth has been reported in other systems with a large lattice misfit, like Zn/ZnO nanostructures.²⁶ Fig. 3a shows a representative HRTEM image of Au nanotriangles on MoS₂ with 1.68 nm-pitch Moiré fringes. Each Moiré fringe represents a misfit dislocation at the Au/MoS₂ interface. Fig. 3b exhibits the dislocations along the [211]-directions of Au or the [2 $\bar{1}$ 0]-directions of MoS₂. Due to the six-fold symmetry of both materials, there are three dislocation arrays with an intersection angle of 120° at the Au/MoS₂ interface. Probably due to the fact that the HRTEM image was not exactly obtained from the [001] zone axis of MoS₂, we can only see one set of Moiré fringes perpendicular to one edge of the nanotriangle. However, we can find the six-fold symmetry of the Moiré fringes by analyzing the Fourier transformation pattern of the TEM image, as shown in the yellow square in Fig. 3c. We also show a schematic (Fig. 3d) that explains the structural relationship between the AuNT and the underlying MoS₂. And, we predict that such a formation mechanism of AuNTs on MoS₂ may govern the growth of other metals with a fcc structure (*e.g.*, Ag, Cu, Ni, and Al) or a hexagonal structure (*e.g.*, Zn), enabling much wider applications. However, for different metals the distance between neighboring dislocations differs as a result of the changed extent of lattice misfit.

Highly-improved Raman enhancement

To search for the possible applications of this kind of MoS₂-based composites, we have used Raman spectroscopy to investigate the SERS properties of well-oriented AuNTs (Fig. 4). The micro-Raman spectroscopy (Renishaw inVia Raman Spectroscopy) experiments were performed under ambient conditions with a 785 nm (1.58 eV) laser. All the spectra were collected under the same ambient conditions (laser power, exposure time, *etc.*). As shown in Fig. 4a and b, it is clearly seen that Raman signals of as-fabricated MoS₂ samples are almost undetectable at 785 nm laser excitation. When a 2 nm gold film is deposited onto the MoS₂ surface at room temperature, we can detect the Raman signal with the Raman peaks appearing clearly, however, the signal is relatively weak. In contrast, if *n*-layer MoS₂ is fully covered by AuNTs prepared by our method, the Raman signals of 1L or 2L MoS₂ are greatly enhanced, indicating that triangular AuNPs can efficiently enhance Raman scattering of *n*-layer MoS₂ compared with ordinary gold particles. This could be attributed to the sharp corners in a nanotriangle, where electromagnetic fields are

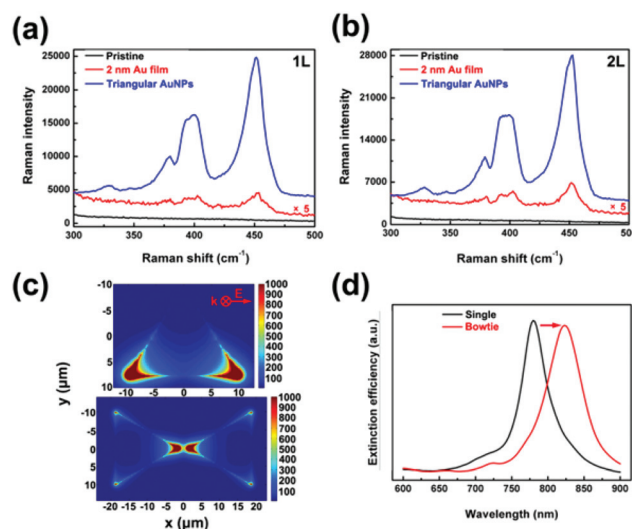


Fig. 4 The comparison among the Raman spectra of original MoS₂, gold film-deposited on MoS₂ and ultrathin AuNTs decorated on 1L (a) and 2L (b) MoS₂ excited by a 785 nm laser. (c) The electromagnetic field enhancement contour external to an individual AuNT and bowtie with a 2 nm thickness and 20 nm edge length at a 785 nm laser excitation. (d) The calculated extinction spectrum of a nanotriangle and relevant bowtie structure with a spacing gap of 3 nm.

concentrated into “hot-spots”, thus resulting in strong Raman enhancements on the Raman signals of MoS₂.^{21,27} In order to figure out the physical origins, we have carried out theoretical simulations (ESI†). Indeed, according to the electromagnetic field contour (Fig. 4c), there is a very strong field enhancement at the sharp edges of a nanotriangle or a bowtie structure. In particular, by examining the relevant extinction spectrum, we can note that there is a strong in-plane dipole plasmon resonance at 785 nm in an AuNT (black in Fig. 4d), while for gold bowties, their extinction spectra also show prominent resonant peaks close to 785 nm (red in Fig. 4d and Fig. S4, ESI†), suggesting the contribution of single AuNT or bowties to the Raman enhancement arising from the coupled surface plasmon resonance absorption. These results would further boost the application of MoS₂-based metal composites possibly in SERS.

Conclusions

In summary, we reported here, the nucleation and growth of equilateral triangular AuNPs on layered MoS₂ with well-defined crystallographic orientations, and an average size controlled by the MoS₂ thickness, suggesting thickness-dependent interactions and the epitaxial growth of Au on *n*-layer MoS₂. Our technique is of great potential for direct synthesis of gold bowtie nanoantennas where two AuNTs opposing tip to tip with the gap as small as 3 nm, which is far below the spatial resolution of e-beam lithography (~8 nm). In particular, these well-oriented AuNTs exhibit excellent Raman enhancing effects, indicating the potential application of AuNTs for molecule detection.

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