

Size, shape, and diffusivity of a single Debye-Hückel polyelectrolyte chain in solution

W. Chamath Soysa, B. Dünweg, and J. Ravi Prakash

Citation: *The Journal of Chemical Physics* **143**, 064906 (2015); doi: 10.1063/1.4928458

View online: <http://dx.doi.org/10.1063/1.4928458>

View Table of Contents: <http://scitation.aip.org/content/aip/journal/jcp/143/6?ver=pdfcov>

Published by the AIP Publishing

Articles you may be interested in

[Interaction between like-charged polyelectrolyte-colloid complexes in electrolyte solutions: A Monte Carlo simulation study in the Debye-Hückel approximation](#)

J. Chem. Phys. **133**, 024901 (2010); 10.1063/1.3459125

[The finite size effect of monomer units on the electrostatics of polyelectrolyte solutions](#)

J. Chem. Phys. **132**, 074904 (2010); 10.1063/1.3322853

[A Brownian dynamics study on the self-diffusion of charged tracers in dilute polyelectrolyte solutions](#)

J. Chem. Phys. **122**, 124905 (2005); 10.1063/1.1869952

[Self-consistent integral equation theory for solutions of finite extensible semiflexible polyelectrolyte chains](#)

J. Chem. Phys. **118**, 6624 (2003); 10.1063/1.1557472

[Integral equation theory of flexible polyelectrolytes. I. Debye-Hückel approach](#)

J. Chem. Phys. **109**, 4659 (1998); 10.1063/1.477071

The cover of the AIP Applied Physics Reviews journal. It features a blue and orange color scheme with a molecular structure background. The text 'AIP Applied Physics Reviews' is at the top left. The main title 'NEW Special Topic Sections' is in large white letters. Below it, 'NOW ONLINE' is in orange, followed by 'Lithium Niobate Properties and Applications: Reviews of Emerging Trends' in white. The AIP Applied Physics Reviews logo is at the bottom right.

NEW Special Topic Sections

NOW ONLINE
Lithium Niobate Properties and Applications:
Reviews of Emerging Trends

AIP Applied Physics Reviews

Size, shape, and diffusivity of a single Debye-Hückel polyelectrolyte chain in solution

W. Chamath Soysa,¹ B. Dünweg,^{1,2,3} and J. Ravi Prakash^{1,a)}

¹Department of Chemical Engineering, Monash University, Melbourne, VIC 3800, Australia

²Max Planck Institute for Polymer Research, Ackermannweg 10, 55128 Mainz, Germany

³Condensed Matter Physics, TU Darmstadt, Karolinenplatz 5, 64289 Darmstadt, Germany

(Received 9 April 2015; accepted 30 July 2015; published online 13 August 2015)

Brownian dynamics simulations of a coarse-grained bead-spring chain model, with Debye-Hückel electrostatic interactions between the beads, are used to determine the root-mean-square end-to-end vector, the radius of gyration, and various shape functions (defined in terms of eigenvalues of the radius of gyration tensor) of a weakly charged polyelectrolyte chain in solution, in the limit of low polymer concentration. The long-time diffusivity is calculated from the mean square displacement of the centre of mass of the chain, with hydrodynamic interactions taken into account through the incorporation of the Rotne-Prager-Yamakawa tensor. Simulation results are interpreted in the light of the Odjik, Skolnick, Fixman, Khokhlov, and Khachaturian blob scaling theory (Everaers *et al.*, Eur. Phys. J. E **8**, 3 (2002)) which predicts that all solution properties are determined by just two scaling variables—the number of electrostatic blobs X and the reduced Debye screening length, Y . We identify three broad regimes, the ideal chain regime at small values of Y , the blob-pole regime at large values of Y , and the crossover regime at intermediate values of Y , within which the mean size, shape, and diffusivity exhibit characteristic behaviours. In particular, when simulation results are recast in terms of blob scaling variables, universal behaviour independent of the choice of bead-spring chain parameters, and the number of blobs X , is observed in the ideal chain regime and in much of the crossover regime, while the existence of logarithmic corrections to scaling in the blob-pole regime leads to non-universal behaviour. © 2015 AIP Publishing LLC. [<http://dx.doi.org/10.1063/1.4928458>]

I. INTRODUCTION

Many synthetic polymers and most biopolymers are polyelectrolytes. Their use in a range of industrial and biological applications makes a thorough understanding of their behaviour highly desirable. While the scaling of many important observable properties of neutral polymers in solution has been successfully predicted by blob theories,^{1,2} much is yet to be understood with this approach even for the simple case of dilute polyelectrolyte solutions at equilibrium. The electrostatic blob, which sets the length scale at which the energy of electrostatic interactions is of order $k_B T$ (where k_B is the Boltzmann constant and T is the temperature), provides the basis for scaling theories of such a system.³ A commonly used scaling theory for polyelectrolyte solutions is the OSFKK scaling picture^{4–7} (after Odjik, Skolnick, Fixman, Khokhlov, and Khachaturian), which predicts the scaling of the mean chain size in different scaling regimes based on the dominating physics. Even though simulations have examined the predictions of OSFKK scaling theory in detail,⁷ many questions remain unanswered and not many studies have applied the blob picture to properties other than the root-mean-square end-to-end vector. The aim of this work is to carry out Brownian dynamics (BD) simulations in order to investigate the dependence of several static and dynamic properties of dilute polyelectrolyte solutions on the intrinsic parameters that govern their behaviour. In particular,

in addition to the mean size, we use the OSFKK scaling picture to interpret simulation predictions of the shape and diffusivity of polyelectrolyte chains. A variety of shape functions, defined in terms of the eigenvalues of the radius of gyration tensor, are used to examine changes in polymer shape in the different regimes of conformational phase space. Further, hydrodynamic interactions (HI) have been incorporated via the Rotne-Prager-Yamakawa tensor,^{8,9} in order to obtain an accurate prediction of the chain diffusivity in various scaling regimes. The OSFKK scaling picture ignores the presence of logarithmic corrections in the blob-pole regime of the conformational phase diagram, which are predicted to be important by more refined scaling theories.¹⁰ For all the properties examined here, we explore whether logarithmic corrections can account for departures from OSFKK scaling in the blob-pole regime.

In most scaling theories, polyelectrolyte molecules are represented as freely jointed chains with N_K Kuhn steps, each of length b_K . The number of charges per Kuhn step, $f = \alpha N_{m,K}$, where $N_{m,K}$ is the number of monomers per Kuhn step, and α is the degree of ionization per chain, is used to characterize the extent of ionic group dissociation. Counterions and salt ions are not modelled explicitly, and the screening of monomer charges due to the presence of free ions is quantified via the Debye screening length, l_D . Another important length scale is the Bjerrum length, l_B , which is the distance at which the Coulomb energy between two unit elementary charges in the solvent is equal to the thermal energy $k_B T$, and is defined by

^{a)}Electronic mail: ravi.jagadeeshan@monash.edu

$$l_B = \frac{e^2}{4\pi\epsilon_0\epsilon_r k_B T}, \quad (1)$$

where e is the charge of an electron, ϵ_0 is the vacuum permittivity, and ϵ_r is the relative dielectric constant of the solvent. Such a representation of a dilute polyelectrolyte solution is referred to here as the bare-model, and in what follows, it is always assumed that the macromolecules are dissolved in a theta solvent (which is simulated by switching off excluded volume interactions).

The electrostatic blob denotes the length scale below which the conformation of a polyelectrolyte chain is practically unaffected by electrostatic interactions, since the electrostatic energy of the sub-chain within a blob is less than the thermal energy. This length scale can be used to divide the chain into a number of blobs, X , of diameter, ξ_{el} , with random walk statistics followed within the blob, while the conformation of the chain of blobs as a whole depends on the electrostatic interactions between the blobs. In addition, within the OSFKK scaling picture, the screening of electrostatic interactions is accounted for by the scaled variable Y , which is the ratio of the Debye length to the blob size. In terms of the bare-model parameters, the OSFKK expressions for the blob scaling variables are⁷

$$X = \frac{N_K}{\hat{\xi}_{el}^2} \quad (2)$$

and

$$Y = \frac{\hat{l}_D}{\hat{\xi}_{el}}, \quad (3)$$

where $\hat{l}_D \equiv l_D/b_K$, and the scaled blob size, $\hat{\xi}_{el} \equiv \xi_{el}/b_K$, is given by

$$\hat{\xi}_{el} = (f^2 \hat{l}_B)^{-\frac{1}{3}} \quad (4)$$

with $\hat{l}_B \equiv l_B/b_K$. Note, for future reference, that it is sufficient to prescribe the reduced set of bare-model parameters $\{N_K, f, \hat{l}_B, \hat{l}_D\}$ in order to determine the scaling variables X and Y .

The OSFKK scaling picture is briefly summarized in Fig. 1, which is a phase diagram in (X, Y) space, highlighting

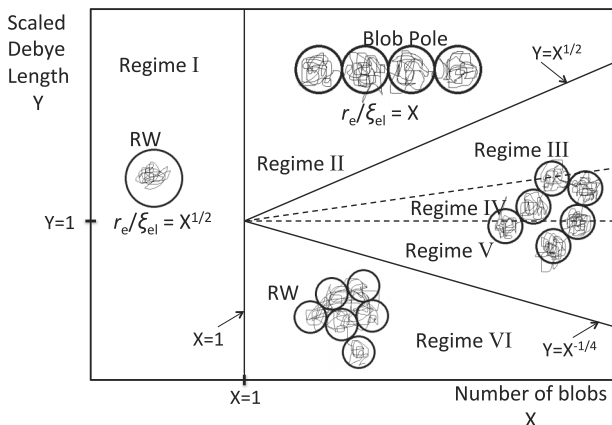


FIG. 1. The phase diagram in the OSFKK scaling picture.⁷ Various regimes are marked I-VI, with scaling predictions for the end-to-end vector r_e , in each regime, listed in Table I. Equations governing the boundaries between regimes are indicated in the figure, and the abbreviation “RW” implies random walk statistics are obeyed.

TABLE I. Scaling expressions for the end-to-end vector in the various regimes shown schematically in Fig. 1, in terms of bare and blob parameters (the value of the Flory excluded volume exponent has been approximated to be 3/5). The variables X and Y are defined in Eqs. (2) and (3), respectively.

Regime	r_e/ξ_{el} (blob parameters)	r_e/b_K (bare parameters)
I	$X^{1/2}$	$N_K^{1/2}$
II	X	$f^{2/3} \hat{l}_B^{1/3} N_K$
III	$Y X^{1/2}$	$f^{2/3} \hat{l}_B^{1/3} \hat{l}_D N_K^{1/2}$
IV	$Y^{3/5} X^{3/5}$	$f^{8/15} \hat{l}_B^{4/15} \hat{l}_D^{3/5} N_K^{3/5}$
V	$Y^{2/5} X^{3/5}$	$f^{2/5} \hat{l}_B^{1/5} \hat{l}_D^{2/5} N_K^{3/5}$
VI	$X^{1/2}$	$N_K^{1/2}$

the different scaling regimes. Table I lists the scaling expressions for the chain size, in terms of both blob and bare parameters, in each of the regimes. In regime I, when $X \ll 1$ (the blob is much larger than the chain), $r_e/\xi_{el} \propto X^{1/2}$, where r_e is the root-mean-square end-to-end vector. The Debye length is irrelevant in this regime as thermal energy completely dominates the electrostatic interactions between the segments. When $X \gg 1$ (when the chain is composed of many electrostatic blobs), the XY plane is divided into different regimes according to how large Y is compared to X . The blob-pole conformation occurs in regime II, when $Y \rightarrow \infty$, and electrostatic interactions between blobs are unscreened. In this regime, $r_e/\xi_{el} \propto X$. When Y decreases below $X^{1/2}$, electrostatic interactions act as short ranged repulsive forces and a crossover regime follows, which, according to Khokhlov and Khachaturian⁶ is decomposed into regimes III-V. Here, the conformations are governed by an intricate interplay of chain stiffness and excluded volume interactions (both of electrostatic origin). Since our numerical investigations lack the resolution to study the details of these regimes, we collectively refer to them as the crossover regime. Finally, when $Y \ll X^{-1/4}$ the Debye length is sufficiently small for the scaling to return to that of an ideal chain.

Everaers *et al.*⁷ have carried out extensive single chain Monte Carlo simulations with Debye-Hückel electrostatic interactions and their results support the OSFKK scaling picture for the end-to-end vector. Pattanayek and Prakash¹¹ used single chain Brownian dynamics simulations with Debye-Hückel electrostatic interactions in order to investigate the scaling of the end-to-end vector and viscometric functions in simple shear flow. They showed that when Brownian dynamics simulation data were represented in terms of blob scaling variables, predictions independent of the choice of parameters in the bead-spring chain model were obtained. As in the present study, these Brownian dynamics simulations were unable to distinguish between the different regimes in the crossover region due to the computational cost of simulating the extremely long chains required to verify OSFKK scaling. By including a characteristic non-dimensional shear rate as an additional scaling variable in shear flow, Pattanayek and Prakash¹¹ were able to show that the equilibrium electrostatic blob model provides a framework to obtain universal scaling even for non-equilibrium properties. Both these simulation studies, however, did not attempt to describe the scaling of other static

properties, or the diffusivity in the different regimes, nor did they examine the occurrence of logarithmic corrections in the blob-pole regime.

The appropriateness of using a Debye-Hückel potential to represent electrostatic interactions in Brownian dynamics simulations has been examined by Stoltz *et al.*¹² They compared the predictions of chain size as a function of Bjerrum length (at various concentrations in the dilute regime) of a bead-spring chain model with explicit counterions and pairwise Coulomb interactions between charges, with those of a model with pair-wise Debye-Hückel electrostatic interactions between beads on chains. They found that the results of both models are nearly identical for values of Bjerrum length roughly equal to or less than the distance between the beads on a chain (all of which were assumed charged). For larger values of Bjerrum length, in accord with Manning's theory,¹³ they observed the onset of counterion condensation. We have adopted a Debye-Hückel potential in this work, and as discussed in greater detail subsequently, chosen parameter values in the bead-spring chain model to ensure that the simulations always remain in the regime where this approximation is valid, and there is no counterion condensation.

Property predictions from Brownian dynamics simulations of bead-spring chains depend on several model parameters, such as the number of beads, the charge on the beads, and the finite extensibility parameter for the springs. A key issue is a rational choice of values for these parameters. The earlier BD simulations of mean size and viscometric functions by Pattanayek and Prakash¹¹ have shown the advantage of using blob scaling variables to interpret results of simulations since this leads to a description independent of the level of coarse-graining, which is extremely useful for comparing simulation results with experiments. In this study, we investigate the shape and diffusivity of polyelectrolyte chains in the various regimes of the phase diagram, by recasting BD simulation results in terms of blob scaling variables, and examine their independence from the specific choice of bead-spring chain model parameters.

Many studies have shown that the shape of a neutral polymer chain, even at equilibrium, is not spherical about the centre of mass of the chain.^{14–30} The nature of the asymmetry in chain shape has been examined in terms of a number of different quantities, such as the *degree of prolateness*, the *asphericity*, the *acylindricity*, the *shape anisotropy*, and so on, which are functions of the eigenvalues of the radius of gyration tensor, since the breaking of symmetry is reflected in the three eigenvalues differing from each other. The symmetry or otherwise of polyelectrolyte chain shapes, particularly in the different scaling regimes, has not yet been systematically investigated. Here, we examine if a universal description of polyelectrolyte chain shapes can be obtained, when BD simulation results are represented in terms of electrostatic blob scaling variables.

While the OSFCK scaling picture ignores the presence of logarithmic corrections to the scaling of chain size with degree of polymerisation in the blob-pole regime, their existence has been derived in a number of different ways, ranging from Flory type energy minimisation arguments¹⁰ to refined scaling theories that account for the nonuniform stretching of polyelectrolyte chains along the elongation axis.^{1,31} In Appendix A

(for the sake of completeness), we have used a Flory type argument to show how logarithmic corrections arise in regime II. In particular, it can be shown that r_e obeys the following scaling expression in terms of the blob scaling variable X :

$$\frac{r_e}{\xi_{el}} \sim X[\ln X]^{\frac{1}{3}}. \quad (5)$$

Liao *et al.*³¹ have carried out molecular dynamic simulations of bead-spring chains with explicit counterions, and have shown in terms of bare-model parameters that for sufficiently long chains, the scaling of the end-to-end vector does indeed exhibit logarithmic corrections in the blob-pole regime. Here, we examine whether results of BD simulations in the blob-pole regime exhibit logarithmic corrections as described by Eq. (5).

A common assumption in the various theoretical descriptions of the blob-pole regime is that electrostatic interactions lead to chain stretching along one direction, while leaving the chain conformation unperturbed in directions perpendicular to the stretching direction.¹⁰ In terms of blob scaling variables, this implies that chain dimensions lateral to the stretching direction are expected to scale as $\xi_{el} X^{\frac{1}{2}}$. By examining the eigenvalues of the gyration tensor, we verify if chain dimensions perpendicular to the stretching direction indeed obey ideal chain scaling laws.

The concept of the Zimm diffusivity of a blob has been successfully used to develop scaling relations for the diffusivity of neutral polymer chains, both in the dilute concentration regime (in terms of thermal blobs²), and in the semidilute regime (in terms of correlation blobs^{2,32}). Here, we examine if the scaling of the diffusivity of polyelectrolyte chains in the various regimes of the phase diagram becomes independent of bead-spring chain parameters, when the Zimm diffusivity of an electrostatic blob is used to interpret simulation results. In particular, since the blob-pole is reminiscent of the shish-kebab model for rodlike polymers³³ (with blobs taking the place of beads), we examine if the diffusivity of a polyelectrolyte chain in regime II can be understood in terms of the translational diffusivity of rodlike polymers.

The paper is structured as follows. In Section II we describe the bead-spring chain model used to represent polyelectrolyte chains, and the governing equations for the time evolution of the position vectors for the beads. Section III, which summarises our results and the relevant discussions, is subdivided into four sections; the first looks at the scaling of the chain size and the extent of logarithmic corrections in the blob-pole regime, the second and third consider the scaling of various functions that describe the shape of the chain, while the fourth examines the scaling of chain diffusivity and relaxation time. Finally, the key findings of this work are summarised in Section IV.

II. THE BEAD-SPRING CHAIN MODEL

A. Governing equations and Brownian dynamics simulations

A polyelectrolyte chain is modelled in the BD simulations by a coarse-grained version of the bare-model, i.e., by a bead-spring chain consisting of N_b beads of radius a , connected

linearly by $(N_b - 1)$ finitely extensible non-linear elastic (FENE) springs, with spring constant H and maximum stretch Q_0 . The beads act as centres of frictional resistance, with a Stokes friction coefficient, $\zeta = 6\pi\eta_s a$ (where η_s is the solvent viscosity). The total charge on the bead-spring chain is set equal to that of a chain in the bare-model, distributed uniformly along the length of the chain with each bead having an identical charge q , given by

$$q = \frac{f N_K}{N_b}. \quad (6)$$

The time evolution of the position vector $\mathbf{r}_\mu(t)$ of bead μ is described by the non-dimensional stochastic differential equation³⁴

$$d\mathbf{r}_\mu = \frac{1}{4} \sum_\nu \mathbf{D}_{\mu\nu} \cdot \mathbf{F}_\nu dt + \frac{1}{\sqrt{2}} \sum_\nu \mathbf{B}_{\mu\nu} \cdot d\mathbf{W}_\nu, \quad (7)$$

where the length scale $l_H = \sqrt{k_B T / H}$ and time scale $\lambda_H = \zeta / 4H$ have been used for non-dimensionalization. The dimensionless diffusion tensor $\mathbf{D}_{\mu\nu}$ is a 3×3 matrix for a fixed pair of beads μ and ν . It is related to the hydrodynamic interaction tensor, as discussed further subsequently. The sum of all the non-hydrodynamic forces on bead ν due to all the other beads is represented by \mathbf{F}_ν , the quantity \mathbf{W}_ν is a Wiener process, and $\mathbf{B}_{\mu\nu}$ is a non-dimensional tensor whose presence leads to multiplicative noise.³⁴ Its evaluation requires the decomposition of the diffusion tensor. Defining the matrices \mathcal{D} and \mathcal{B} as block matrices consisting of $N \times N$ blocks each having dimensions of 3×3 , with the (μ, ν) -th block of \mathcal{D} containing the components of the diffusion tensor $\mathbf{D}_{\mu\nu}$, and the corresponding block of \mathcal{B} being equal to $\mathbf{B}_{\mu\nu}$, the decomposition rule for obtaining \mathcal{B} can be expressed as

$$\mathcal{B} \cdot \mathcal{B}^T = \mathcal{D}. \quad (8)$$

The non-hydrodynamic forces on a bead μ are comprised of the non-dimensional spring forces \mathbf{F}_μ^s and non-dimensional electrostatic interaction forces $\mathbf{F}_\mu^{\text{es}}$, i.e., $\mathbf{F}_\mu = \mathbf{F}_\mu^s + \mathbf{F}_\mu^{\text{es}}$. The entropic spring force on bead μ due to adjacent beads can be expressed as $\mathbf{F}_\mu^s = \mathbf{F}^c(\mathbf{Q}_\mu) - \mathbf{F}^c(\mathbf{Q}_{\mu-1})$ where $\mathbf{F}^c(\mathbf{Q}_{\mu-1})$ is the force between the beads $\mu - 1$ and μ , acting in the direction of the connector vector between the two beads $\mathbf{Q}_{\mu-1} = \mathbf{r}_\mu - \mathbf{r}_{\mu-1}$. Specifically, the spring force in the FENE springs used here is given by

$$\mathbf{F}^c(\mathbf{Q}_\mu) = \frac{\mathbf{Q}_\mu}{1 - (\mathbf{Q}_\mu^2/b)}, \quad (9)$$

where $b = H Q_0^2 / k_B T$ is the dimensionless finite extensibility parameter. The vector $\mathbf{F}_\mu^{\text{es}}$ is given in terms of the dimensionless electrostatic potential $U_{\mu\nu}^{\text{es}}$ between the beads μ and ν of the chain,

$$\mathbf{F}_\mu^{\text{es}} = - \sum_{\substack{\nu=1 \\ \nu \neq \mu}}^{N_b} \frac{\partial}{\partial \mathbf{r}_\mu} U_{\mu\nu}^{\text{es}}. \quad (10)$$

The OSFKK scaling theory assumes Debye-Hückel electrostatic interactions³⁵ between charges, which is justified for weakly charged polyelectrolyte chains in the absence of Manning counterion condensation.^{12,13,36,37} We adopt a

Debye-Hückel potential in this work,

$$U_{\mu\nu}^{\text{es}} = \frac{l_B^* q^2}{r_{\mu\nu}} \exp\left(-\frac{r_{\mu\nu}}{l_D^*}\right) \quad (11)$$

with $r_{\mu\nu} = |\mathbf{r}_{\mu\nu}|$, where $\mathbf{r}_{\mu\nu} = \mathbf{r}_\mu - \mathbf{r}_\nu$ is the vector between beads ν and μ , and l_B^* and l_D^* are the non-dimensional Bjerrum and Debye lengths, respectively.

The non-dimensional diffusion tensor $\mathbf{D}_{\mu\nu}$ is related to the non-dimensional hydrodynamic interaction tensor $\mathbf{\Omega}$ through

$$\mathbf{D}_{\mu\nu} = \delta_{\mu\nu} \delta + (1 - \delta_{\mu\nu}) \mathbf{\Omega}(\mathbf{r}_\nu - \mathbf{r}_\mu), \quad (12)$$

where δ and $\delta_{\mu\nu}$ represent a unit tensor and a Kronecker delta, respectively, while $\mathbf{\Omega}$ represents the effect of the motion of a bead μ on another bead ν through the disturbances carried by the surrounding fluid. The hydrodynamic interaction tensor $\mathbf{\Omega}$ is assumed to be given by the RPY regularisation of the Oseen function

$$\mathbf{\Omega}(\mathbf{r}) = \Omega_1 \delta + \Omega_2 \frac{\mathbf{r}\mathbf{r}}{r^2}, \quad (13)$$

where for $r \equiv |\mathbf{r}| \geq 2a^*$,

$$\Omega_1 = \frac{3}{4} \frac{a^*}{r} \left(1 + \frac{2}{3} \frac{a^{*2}}{r^2}\right) \quad \text{and} \quad \Omega_2 = \frac{3}{4} \frac{a^*}{r} \left(1 - 2 \frac{a^{*2}}{r^2}\right) \quad (14)$$

while for $0 < r \leq 2a^*$,

$$\Omega_1 = 1 - \frac{9}{32} \frac{r}{a^*} \quad \text{and} \quad \Omega_2 = \frac{3}{32} \frac{r}{a^*}. \quad (15)$$

Here, a^* is the non-dimensional bead radius, which is related to the conventionally defined³⁸ hydrodynamic interaction parameter, h^* , by $a^* = \sqrt{\pi} h^*$. We have set $a^* = 0.5$ in all simulations in which hydrodynamic interactions are incorporated. A choice of h^* close to 0.25 ensures that the non-draining limit is reached (and consequently universal predictions obtained), at relatively small values of N_b .³⁹⁻⁴⁴

The spatial configuration of the chain at any time t , i.e., $\mathbf{r}_\mu(t)$ for all beads $\mu = 1, \dots, N_b$, is obtained by integrating Eq. (7) using a semi-implicit predictor-corrector scheme proposed by Prabhakar and Prakash.⁴⁵ In the presence of fluctuating HI, the problem of the computational intensity of calculating the Brownian term is reduced by the use of a Chebyshev polynomial representation for the Brownian term.^{46,47} We have adopted this strategy, and the details of the exact algorithm followed here are given in Ref. 45.

While the majority of results reported here have been obtained with the “single-chain” BD algorithm described in Ref. 45, the predictions of eigenvalues and shape functions have been obtained with the “multi-chain” BD algorithm described in Ref. 48. The most significant difference between the two algorithms is that the latter (in which multiple chains are simulated in a box with periodic boundary conditions) accounts for *inter-particle* hydrodynamic and electrostatic interactions in addition to *intra-particle* interactions. Essentially, N_c bead-spring chains are simulated in a cubic simulation box of length L , such that the concentration $c = N_b N_c / L^3$. With the overlap concentration c^* defined by $c^* = N_b / [(4/3)\pi r_{g,\text{eq}}^3]$, where $r_{g,\text{eq}}$ is the radius of gyration of a chain at equilibrium, we maintain a scaled concentration $c/c^* = 10^{-5}$ to ensure that the system is in the dilute limit. Since the typical distance between chains at these concentrations is much greater than

the Debye screening lengths considered here, the strength of short-ranged inter-molecular Debye-Hückel electrostatic interactions is effectively zero. The equivalence of the use of either of the algorithms has been verified by comparison of predictions of the end-to-end vector and the radius of gyration, and ensuring that no difference was observed. Plots which contain data from the multi-chain BD algorithm are identified as such in the figure captions, and unless explicitly stated, most plots report data obtained using the single chain BD algorithm.

B. Size, shape, and diffusivity

The two static properties examined here are: (i) the end-to-end distance, $r_e \equiv \sqrt{\langle r_e^2 \rangle}$, with

$$\langle r_e^2 \rangle = \langle (\mathbf{r}_{N_b} - \mathbf{r}_1) \cdot (\mathbf{r}_{N_b} - \mathbf{r}_1) \rangle, \quad (16)$$

where $\langle (\cdot) \rangle$ represents an ensemble average, and \mathbf{r}_{N_b} and \mathbf{r}_1 are the positions of the two beads at either end of the chain and (ii) the radius of gyration of the chain, $r_g \equiv \sqrt{\langle r_g^2 \rangle}$, with

$$\langle r_g^2 \rangle = \langle \lambda_1^2 \rangle + \langle \lambda_2^2 \rangle + \langle \lambda_3^2 \rangle, \quad (17)$$

where λ_1^2 , λ_2^2 , and λ_3^2 are eigenvalues of the gyration tensor \mathbf{G} (arranged in ascending order), with

$$\mathbf{G} = \frac{1}{2N_b^2} \sum_{\mu=1}^{N_b} \sum_{\nu=1}^{N_b} \mathbf{r}_{\mu\nu} \mathbf{r}_{\mu\nu}. \quad (18)$$

Note that, \mathbf{G} , λ_1^2 , λ_2^2 , and λ_3^2 are calculated for each trajectory in the simulation before the ensemble averages are evaluated.

The asymmetry in equilibrium chain shape has been studied previously in terms of various functions defined in terms of the eigenvalues of the gyration tensor.^{14–30} Apart from λ_1^2 , λ_2^2 , and λ_3^2 , themselves, we have examined the following *shape functions*: the asphericity (B), the acylindricity (C), the degree of prolateness (S and S^*), and the shape anisotropy (κ^2 and κ^{2*}), as defined in Table II. As is evident from the table, the latter two quantities are evaluated in terms of two different definitions. Typically, it is easier to evaluate S and κ^2 for analytical calculations, rather than S^* and κ^{2*} , which require the evaluation of averages of ratios of fluctuating quantities.³⁰

TABLE II. Definitions of shape functions in terms of eigenvalues of the gyration tensor, \mathbf{G} . Note that $I_1 = \lambda_1^2 + \lambda_2^2 + \lambda_3^2$ and $I_2 = \lambda_1^2 \lambda_2^2 + \lambda_2^2 \lambda_3^2 + \lambda_3^2 \lambda_1^2$ are invariants of \mathbf{G} .

Shape function	Definition	
Asphericity ²⁰	$B = \langle \lambda_3^2 \rangle - \frac{1}{2} [\langle \lambda_1^2 \rangle + \langle \lambda_2^2 \rangle]$	(19)
Acylindricity ²⁰	$C = \langle \lambda_2^2 \rangle - \langle \lambda_1^2 \rangle$	(20)
Degree of prolateness ^{22,27}	$S^* = \left\langle \frac{(3\lambda_1^2 - I_1^2)(3\lambda_2^2 - I_1^2)(3\lambda_3^2 - I_1^2)}{(I_1^2)^3} \right\rangle$	(21)
	$S = \frac{\langle (3\lambda_1^2 - I_1^2)(3\lambda_2^2 - I_1^2)(3\lambda_3^2 - I_1^2) \rangle}{\langle (I_1^2)^3 \rangle}$	(22)
Relative shape anisotropy ^{20,22,27}	$\kappa^{2*} = 1 - 3 \left\langle \frac{I_2}{I_1^2} \right\rangle$	(23)
	$\kappa^2 = 1 - 3 \frac{\langle I_2 \rangle}{\langle I_1^2 \rangle}$	(24)

The only dynamic property that is directly calculated here is the long-time self-diffusion coefficient D , defined by

$$D = \lim_{t \rightarrow \infty} \left\langle \frac{|\mathbf{r}_c(t) - \mathbf{r}_c(0)|^2}{6t} \right\rangle, \quad (25)$$

where $\mathbf{r}_c = \frac{1}{N_b} \sum_{\mu=1}^{N_b} \mathbf{r}_\mu$ is the position vector of the center of mass of the chain.

The calculation of the radius of gyration and the diffusivity of a chain enables an estimation of the longest relaxation time from the expression

$$\tau = \frac{\langle r_g^2 \rangle}{D}. \quad (26)$$

An ensemble averaging method has been used to estimate all the properties, details of which are given in Appendix A of Ref. 49. We find that trajectories level off after roughly one relaxation time, but a true stationary state from which equilibrium properties can be estimated generally occurs after tens of relaxation times. Typically, more than $O(10^6)$ data points are used in the estimation of averages and standard errors of mean, where a data point is a saved property value at some time after the beginning of the stationary state. For properties evaluated with the single chain code, a time step size of 0.005 was used, and data were saved roughly after every 200 time steps. Averages were carried out over 1000 trajectories, with 1000 data points in each trajectory. For properties evaluated with the multi-chain code, a time step size of 0.005 was used, and data were saved roughly after every 10 to 20 time steps. Averages were carried out over 64 trajectories, each with 15 chains in the simulation box, and 2000 data points in each trajectory.

C. Mapping between bead-spring chain and blob variables

The concept of an electrostatic blob enables the representation of experimental and simulation data for dilute polyelectrolyte solutions in terms of a significantly reduced number of scaling variables ($\{X, Y\}$), when compared to the number of bare-model parameters ($\{N_K, f, \hat{l}_B, \hat{l}_D\}$). Further, the universal nature of property predictions when expressed in terms of blob scaling variables has been established previously by simulations, which have shown that various combinations of bare-model parameters lead to identical results, provided values of the scaling variables are kept fixed.⁷ In order to achieve a similar demonstration for BD simulations of bead-spring chains, it is necessary to map the bead-spring chain parameters $\{N_b, b, q, l_B^*, l_D^*\}$ onto the blob scaling variables $\{X, Y\}$. This issue has been discussed previously by Pattanayek and Prakash.¹¹ For the sake of completeness, we summarise their arguments here.

The key first step is to map bead-spring chain parameters onto bare-model parameters, after which the mapping onto blob scaling variables is straight-forward. This is achieved by equating the end-to-end vectors and the contour lengths of the bare-model and bead-spring chains. If L is the contour length, then

$$L = N_K b_K = (N_b - 1) Q_0 = (N_b - 1) \sqrt{b} l_H. \quad (27)$$

Further, using the analytical result for the end-to-end vector of FENE chains,³⁸ we get

$$\langle r_e^2 \rangle = N_K b_K^2 = (N_b - 1) \frac{3b}{b+5} l_H^2. \quad (28)$$

Equations (27) and (28) can be solved for N_K and $b_K^* = b_K/l_H$ to give

$$N_K = (N_b - 1) \frac{b+5}{3}, \quad b_K^* = \frac{3\sqrt{b}}{b+5}. \quad (29)$$

Since $\hat{l}_D = l_D^*/b_K^*$, $\hat{l}_B = l_B^*/b_K^*$, and $\xi_{el} = \xi_{el}^*/b_K^*$, substituting from Eq. (29) for N_K and b_K^* into Eqs. (2) and (3), and using Eq. (6), leads to the following equations for the blob scaling variables in terms of bead-spring chain parameters:

$$X = (N_b - 1) \left[\frac{3b}{b+5} \right] \frac{1}{\xi_{el}^{*2}}, \quad (30)$$

$$Y = \frac{l_D^*}{\xi_{el}^*}, \quad (31)$$

where

$$\xi_{el}^* = \left[\frac{9(N_b - 1)^2 b^2}{N_b^2 q^2 l_B^* (b+5)^2} \right]^{1/3}. \quad (32)$$

It is clear from the expressions above that various combinations of bead-spring chain parameters can result in identical values for X and Y . Table III displays the various sets of values of bead-spring chain parameters, and the resultant sets of scaling variables X and Y , that have been used in the simulations reported here, and the corresponding bare-model parameter values. Two to three different values of N_b are typically used for each value of X . Note that Y can be varied independent of X by varying l_D^* (and keeping all other bead-spring chain parameters fixed). As a result, the dependence of properties of dilute polyelectrolyte solutions on X and Y can be explored by carrying out BD simulations, and their extent of universality assessed.

In terms of bare-model parameters, the Manning parameter, γ_0 , which is defined as the number of elementary charges

per Bjerrum length,¹⁰ is given by $\gamma_0 = f\hat{l}_B$. The OSFCK scaling theory assumes that $\gamma_0 \ll 1$, since counterion condensation, which occurs for $\gamma_0 \gtrsim 1$, is not taken into account.^{10,12} The last column in Table III shows that the Manning parameter is significantly smaller than one in all the simulations reported here.

D. Mapping from experiments to blob variables

While the results of the simulations reported here have not been compared directly with experimental observations, it is worth outlining how experimental data on dilute polyelectrolyte solutions can be mapped onto blob variables, to facilitate future comparisons. In this context, the various equations for evaluating the blob variables X and Y for an arbitrary polyelectrolyte solution are derived in Appendix B.

In the current simulation, the behaviour of various static properties and the diffusivity of dilute polyelectrolyte solutions has been explored for a range of blob variable values, $20 < X < 80$ and $0.01 \leq Y \leq 100$. In Appendix B, the specific example of an aqueous solution of NaPSS is considered, and it is shown how an appropriate choice of temperature, polymer molecular weight, degree of ionization per chain, and salt and monomer concentration can lead to blob variables that span this range. Further, parameter values are chosen to ensure that the Manning parameter $\gamma_0 \lesssim 1$ in order to avoid counterion condensation.

III. RESULTS

A. Mean chain size as a function of X and Y

From the OSFCK scaling picture depicted schematically in Fig. 1, we expect the chain size to be independent of Y in the limits of $Y \rightarrow 0$ (regime VI) and $Y \rightarrow \infty$ (regime II) for $X \gg 1$, while at intermediate values of Y (in the crossover regimes III, IV, and V), we expect the chain size to depend on Y . This expectation has been verified by several prior simulations,

TABLE III. Values for the bead-spring chain parameters $\{N_b, b, q, l_B^*\}$, the resultant scaling variables X , and ξ_{el}^* , and the corresponding values for the bare model parameters $\{N_K, f, \hat{l}_B\}$, used in Brownian dynamics simulations. Note that the listed values of the scaling variable Y are specified independent of other parameters by suitably choosing l_D^* (not tabulated here). L^* is the non-dimensional contour length of the bead-spring chain, and γ_0 is the Manning parameter.

Blob model			Bead-spring chain model		Bare-model	
X	Y	ξ_{el}^*	$\{N_b, b, q, l_B^*\}$	L^*	$\{N_K, f, \hat{l}_B\}$	γ_0
20.3	30	2.07	{32, 72.42, 1.1, 0.69}	272	{800, 0.044, 2.09}	0.092
24	0.01, 0.05, 0.1, 0.5, 1, 1.6, 3, 4.8, 10, 20, 30, 100	1.77	{28, 61.67, 1.79, 0.41}	220	{600, 0.0833, 1.15}	0.096
		1.90	{32, 72.42, 1.25, 0.69}	272	{800, 0.05, 2.08}	0.104
		2.21	{40, 61.67, 1.38, 0.4}	314	{867, 0.0638, 1.13}	0.072
27.6	30	1.77	{32, 61.67, 1.79, 0.41}	251	{689, 0.0831, 1.16}	0.096
30.74		1.68	{32, 72.42, 1.5, 0.69}	272	{800, 0.06, 2.09}	0.182
34.83		1.72	{38, 72.42, 1.45, 0.69}	323	{955, 0.0577, 2.09}	0.170
37.67		1.70	{40, 72.42, 1.48, 0.69}	340	{1006, 0.0588, 2.09}	0.125
48	0.01, 0.05, 0.1, 0.5, 1, 1.6, 3, 4.8, 10, 20, 30, 100	1.73	{52, 72, 1.81, 0.45}	441	{1310, 0.0718, 1.36}	0.098
		1.85	{60, 61.67, 1.70, 0.41}	471	{1310, 0.0777, 1.16}	0.090
72	0.01, 0.05, 0.1, 0.5, 1, 1.6, 3, 4.8, 10, 20, 30, 100	1.70	{75, 72, 1.95, 0.41}	636	{1900, 0.0771, 1.24}	0.096
		1.80	{85, 61.67, 1.61, 0.5}	668	{1900, 0.0732, 1.41}	0.103

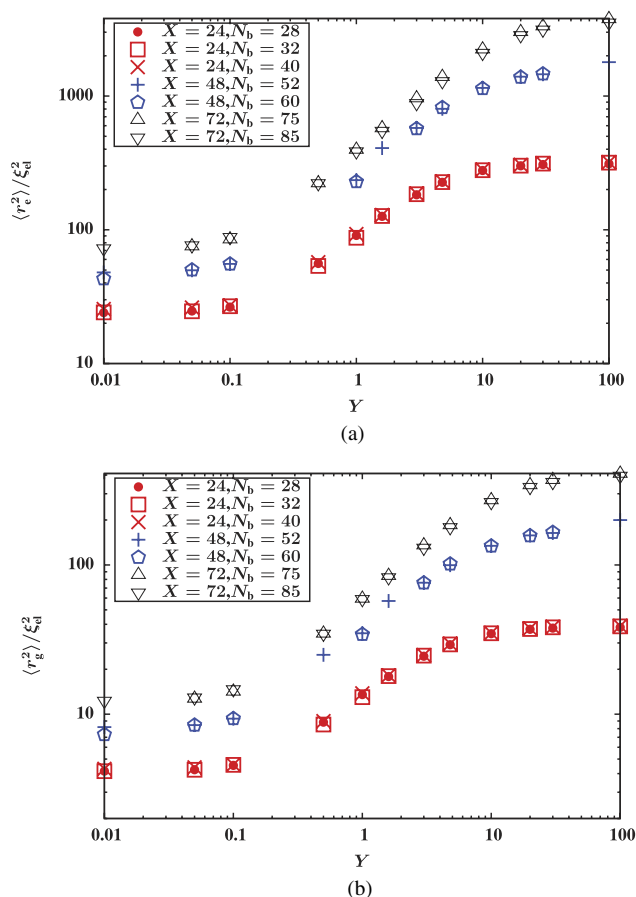


FIG. 2. Measures of mean chain size versus the reduced screening length, Y , at various values of the number of blobs, X : (a) scaled end-to-end vector, $\langle r_e^2 \rangle / \xi_{el}^2$. (b) scaled radius of gyration, $\langle r_g^2 \rangle / \xi_{el}^2$. Simulation data acquired at different values of chain length, N_b , are displayed at each value of X , to demonstrate parameter-free data collapse. The various combinations of bead-spring chain parameters that correspond to the displayed values of N_b are shown in Table III. Error bars are of the order of symbol size.

which have examined the scaling of the end-to-end vector.^{7,11,31} Figs. 2(a) and 2(b) show that in the current simulations as well, both the scaled end-to-end vector, $\langle r_e^2 \rangle / \xi_{el}^2$, and the scaled radius of gyration, $\langle r_g^2 \rangle / \xi_{el}^2$, asymptotically approach constant values at the two extremes of Y . Note that the asterisk in ξ_{el}^{*2} has been dropped for simplicity of notation, since the non-dimensional character of the blob size is clear from the context. In the crossover regime, the chain size increases monotonically with Y as the chain starts to swell due to increasing electrostatic repulsion with increasing Debye screening length. As mentioned previously, our BD simulations are unable to distinguish between the different regimes in the crossover region due to the computational cost of simulating the extremely long chains that would be required for such a purpose.

Notably, as long as the value of X is the same, the predicted values of both $\langle r_e^2 \rangle / \xi_{el}^2$ and $\langle r_g^2 \rangle / \xi_{el}^2$ are independent of the specific choice of bead-spring chain parameters, at all values of Y . For instance at $X = 48$, both $N_b = 52$ and $N_b = 60$ lead to identical results. Such a parameter-free data collapse was previously shown to occur for BD simulation predictions of $\langle r_e^2 \rangle / \xi_{el}^2$ by Pattanayek and Prakash,¹¹ who also noted that the only constraint to the simulations appeared to be to ensure that there were more beads than blobs in a chain, i.e., $N_b \geq X$.

The OSFKK scaling picture ignores the presence of logarithmic corrections in the blob-pole regime. However, as seen in Appendix A, and as shown previously by several different analytical studies,^{1,10} such corrections are expected to arise in this regime, and previous molecular simulations have shown that such is indeed the case.³¹ In the absence of logarithmic corrections, the scaled end-to-end vector, $\langle r_e^2 \rangle / \xi_{el}^2$, is expected to scale as X^2 in regime II (see Table I). It is clear from Fig. 3(a) that while this scaling is obeyed for values of $X \lesssim 30$, there is a departure at larger values of X . A similar deviation from the expected scaling in regime II was observed previously in the BD simulations reported by Pattanayek and Prakash.¹¹ However, they did not verify if the source of the deviation was due to logarithmic corrections. In terms of the blob scaling variable X , Eq. (5) shows that the logarithmic correction term has an exponent of $1/3$. We expect, therefore, that a plot of $r_e / (\xi_{el} X)$ versus $\ln X$ would be a straight line with a slope of $1/3$. Fig. 3(b) shows that the logarithmic corrections to OSFKK scaling, picked up by the BD simulations, do in fact have the expected dependence on X .

While the radius of gyration is also a measure of mean chain size, Fig. 4 reveals that the logarithmic corrections to

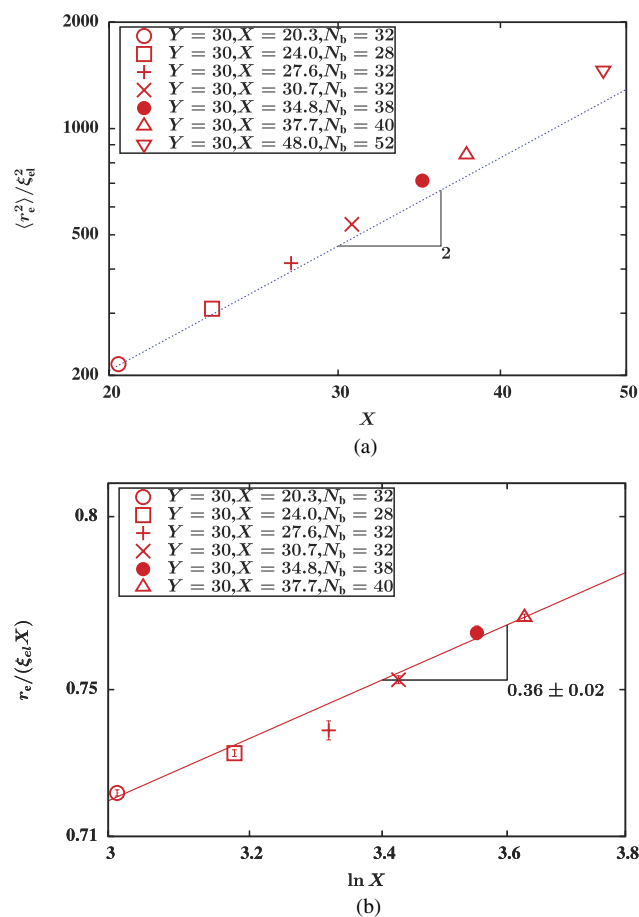


FIG. 3. Demonstration of the existence of logarithmic corrections to scaling at $Y = 30$ (in the blob-pole regime) for the scaled end-to-end vector. (a) The dotted line shows that $\langle r_e^2 \rangle / \xi_{el}^2$ departs from the X^2 scaling predicted by OSFKK theory for $X \gtrsim 30$. (b) A power law fit through the data for $r_e / (\xi_{el} X)$ versus $\ln X$ (solid line) shows that the logarithmic corrections have the expected exponent of $1/3$ (within error bars). The various combinations of bare parameters corresponding to the value of N_b are shown in Table III. Error bars are of the order of symbol size.

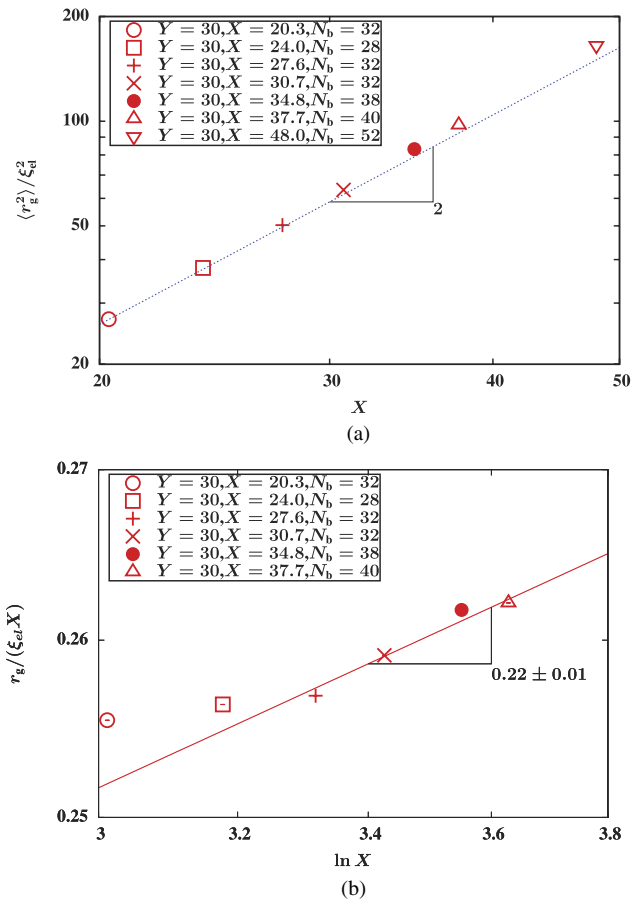


FIG. 4. Demonstration of the existence of weak logarithmic corrections to scaling at $Y = 30$ (in the blob-pole regime) for the scaled radius of gyration. (a) The dotted line shows the onset of departure from X^2 scaling for $\langle r_g^2 \rangle / \xi_{el}^2$ at large values of X . (b) A power law fit through the last four data points for $r_g / (\xi_{el} X)$ versus $\ln X$ (solid line) shows that the logarithmic corrections have an exponent smaller than $1/3$. The various combinations of bare parameters corresponding to the value of N_b are shown in Table III. Error bars are of the order of symbol size.

scaling for $\langle r_g^2 \rangle / \xi_{el}^2$ in the blob-pole regime are much weaker than for $\langle r_e^2 \rangle / \xi_{el}^2$. This could be attributed to the fact that the radius of gyration not only depends on the stretched axis of the chain (which is expected to be in the direction of the end-to-end vector), but also depends on the dimensions of the chain perpendicular to the elongation axis. The scaling of the chain in these lateral dimensions is generally considered to obey random walk statistics. We examine the validity of this expectation below, when we consider the eigenvalues of the gyration tensor.

In the limit of long chains, the ratio of the mean square end-to-end vector to the mean square radius of gyration, $\langle r_e^2 \rangle / \langle r_g^2 \rangle$, has a universal value of 6 for ideal chains, while it is equal to 12 for rigid rods.^{2,33} We anticipate, therefore, that for the polyelectrolyte solutions considered here, the value of the ratio will increase from being close to 6 in the limit $Y \ll 1$ (where the chain behaves ideally) towards a value of 12, as $Y \rightarrow \infty$, where the polyelectrolyte chain adopts a stretched rodlike configuration. Fig. 5(a) shows that the ratio does approach a constant value close to 6 as Y approaches zero, and increases monotonically with increasing Y . However, though the ratio levels off as the chain enters the blob-pole regime in

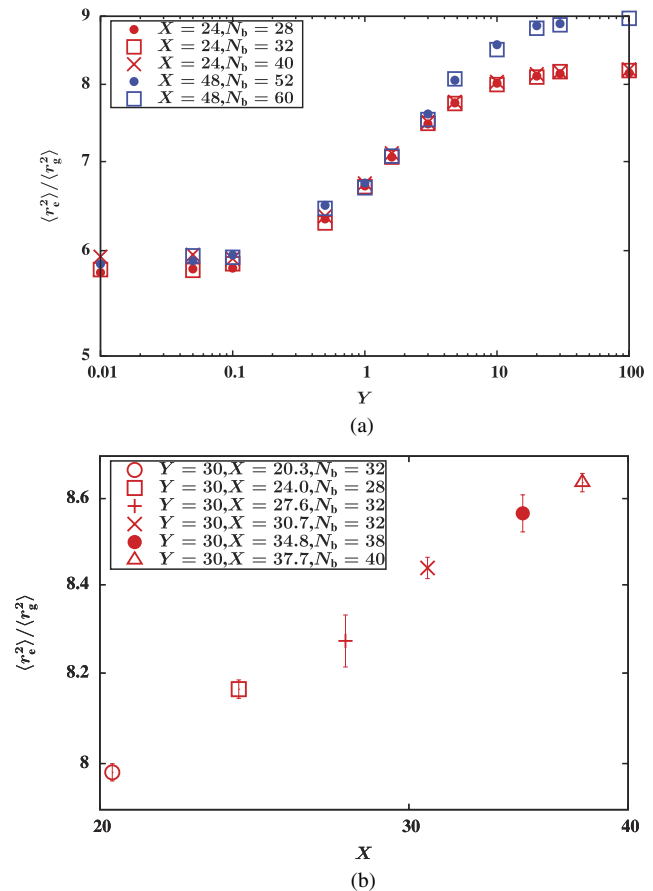


FIG. 5. The variation of the ratio $\langle r_e^2 \rangle / \langle r_g^2 \rangle$ with (a) the reduced screening length Y , for two different values of the number of blobs X , and (b) the number of electrostatic blobs X , at $Y = 30$. Error bars are of the order of symbol size in (a), and are explicitly displayed in (b).

the limit of large Y , the asymptotic value is less than 12, for the values of X considered here. The departure from the limiting value for rigid rods can be considered to reflect the degree of flexibility remaining in the chain in this regime. Interestingly, the ratio appears to be independent of X for a range of values of Y , starting with the ideal regime, and progressing well into the crossover regime (i.e., the red and blue symbols overlap in Fig. 5(a)). The ratio starts to become X dependent for values of Y approaching the blob-pole regime. As was seen in Figs. 3(b) and 4(b) above, though both $\langle r_e^2 \rangle$ and $\langle r_g^2 \rangle$ depart from the OSFKK scaling theory predictions in this regime due to the presence of logarithmic corrections, the strength of these corrections is different in the two cases. This difference is responsible for a persistent dependence of $\langle r_e^2 \rangle / \langle r_g^2 \rangle$ on X in the blob-pole regime, as can be seen in Fig. 5(b), where the value of the ratio is plotted as a function of X , at $Y = 30$. We can anticipate, however, that with increasing values of X (i.e., chain length), the appearance of logarithmic corrections, and the consequent onset of non-universal behaviour, is postponed to larger and larger values of Y , due to a delay in the inception of the blob-pole regime (see Fig. 1).

B. Eigenvalues of the radius of gyration tensor

The dependence of the three scaled eigenvalues of the radius of gyration tensor, $\langle \lambda_1^2 \rangle / \xi_{el}^2$, $\langle \lambda_2^2 \rangle / \xi_{el}^2$, and $\langle \lambda_3^2 \rangle / \xi_{el}^2$, on

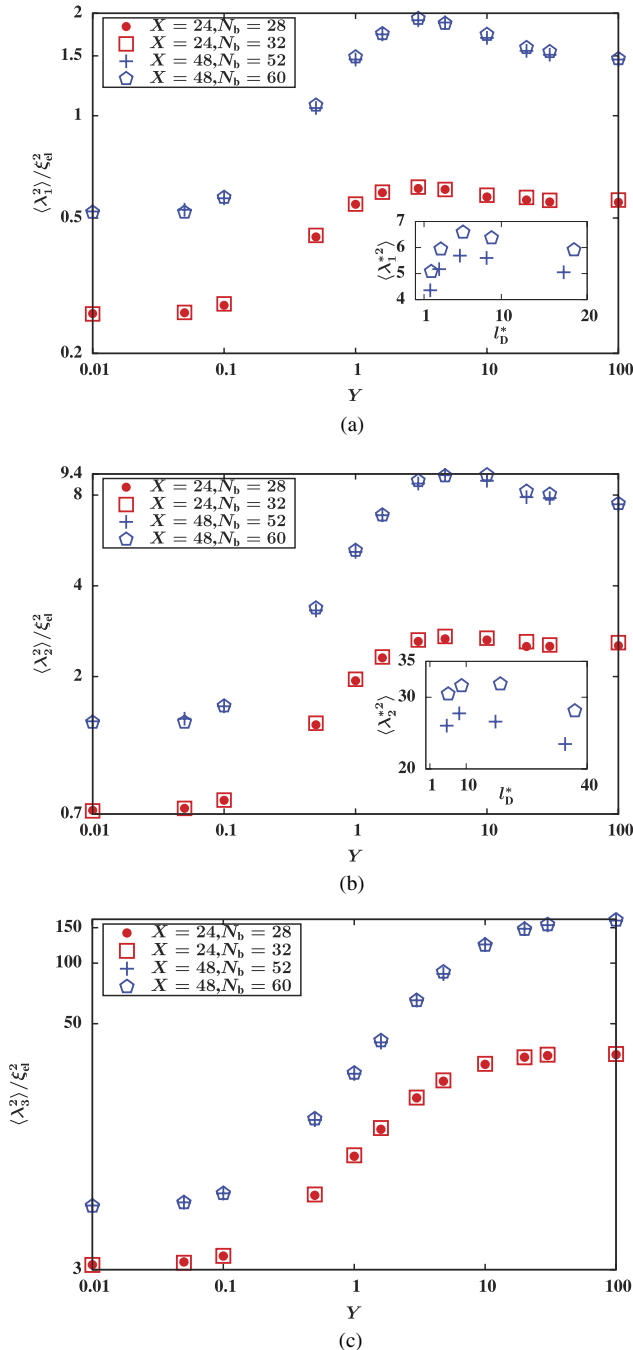


FIG. 6. Scaled eigenvalues of the gyration tensor as a function of Y , at two values of X : (a) the smallest eigenvalue $\langle \lambda_1^2 \rangle / \xi_{el}^2$, (b) the intermediate eigenvalue $\langle \lambda_2^2 \rangle / \xi_{el}^2$, and (c) the largest eigenvalue $\langle \lambda_3^2 \rangle / \xi_{el}^2$. The insets in (a) and (b) display the dependence of the unscaled eigenvalues on the non-dimensional Debye length l_D^2 close to the overshoot, for two different chain lengths, N_b , at the same value of X . Data obtained with the multi-chain algorithm, with $c/c^* = 10^{-5}$. Error bars are of the order of symbol size.

the reduced screening length Y , is shown in Fig. 6. As was observed for $\langle r_e^2 \rangle / \xi_{el}^2$ and $\langle r_g^2 \rangle / \xi_{el}^2$, all the eigenvalues increase in the crossover regime from constant values at small values of Y , to constant values in the blob-pole regime at large values of Y . The collapse of data for different bead-spring chain model parameters, when represented in terms of blob scaling variables, is also observed for these properties. Clearly, the chain size increases with increasing electrostatic repulsion between

blobs, not only in the direction of maximum chain stretching ($\langle \lambda_3^2 \rangle$), but also in the lateral directions ($\langle \lambda_1^2 \rangle$ and $\langle \lambda_2^2 \rangle$). The significant differences in the magnitude of the chain dimensions in the three different principal directions clearly illustrates the highly anisotropic shape of the chain in all regions of the phase diagram. This is examined in more detail in terms of shape functions in Sec. III C. A curious observation, for which we do not have an obvious explanation, is the presence of an overshoot in $\langle \lambda_1^2 \rangle / \xi_{el}^2$ and $\langle \lambda_2^2 \rangle / \xi_{el}^2$ for values of Y at the threshold of the blob-pole regime, i.e., just before the chain enters the final screening-length-independent regime from the crossover regime. The continued stretching of $\langle \lambda_3^2 \rangle$ with increasing Y at the verge of the blob-pole regime appears to be accommodated by a shrinkage in $\langle \lambda_1^2 \rangle$ and $\langle \lambda_2^2 \rangle$. The overshoot also becomes more pronounced with increasing X . The insets in Figs. 6(a) and 6(b) demonstrate that even the occurrence of an overshoot, which is distinct for different bead-spring chain parameters, collapses onto a single curve when represented in terms of scaling variables—signifying that blob variables accurately capture the essential physics, even for phenomena that go beyond the simple picture upon which they were originally based. The use of finitely extensible springs in the simulations is probably not responsible for the overshoot, and it is more likely to be an electrostatic phenomenon, since the springs are weakly stretched even in the blob-pole regime. For instance, from the value of $\langle r_e^2 \rangle / \xi_{el}^2$ in Fig. 2(a) ($\langle r_e^2 \rangle / \xi_{el}^2 \approx 2000$, for $N_b = 60$ and $X = 48$, in the limit of large Y), and the values of ξ_{el}^* and L^* corresponding to this value of N_b and X in Table III ($\xi_{el}^* = 1.85$ and $L^* = 471$), we can see that the ratio of the end-to-end vector to the contour length is less than 16%.

The scaling of r_e with X in the blob-pole regime was seen in Fig. 3(b) to exhibit logarithmic corrections in line with the prediction of Eq. (5). Since the largest eigenvalue of the radius of gyration tensor is expected to be in the direction of maximum chain stretching, we expect $\sqrt{\langle \lambda_3^2 \rangle} / \xi_{el}$ to also exhibit logarithmic corrections according to Eq. (5). Figure 7(a) displays the departure from the OSFKK scaling prediction in the blob-pole regime, while Fig. 7(b) shows that the logarithmic corrections have an exponent close to the expected value of $1/3$.

According to OSFKK theory, the scaling of chain dimensions lateral to the elongation axis is expected to remain unperturbed in the blob-pole regime, i.e., to obey random walk statistics. To our knowledge, there are no theories that describe modifications to this scaling behaviour due to the nonuniform stretching of the chain (unlike the refined theories developed for scaling in the direction of maximum stretching). The dependence of chain dimensions on the number of blobs, in the directions perpendicular to the stretching direction, is examined in Figs. 8(a) and 8(b), where $\langle \lambda_1^2 \rangle / \xi_{el}^2$ and $\langle \lambda_2^2 \rangle / \xi_{el}^2$ are plotted as functions of X . For the limited range of values of X examined here, the smallest and intermediate eigenvalues appear to obey power law scaling with exponents that lie somewhere between ideal chain and blob-pole scaling.

Eigenvalues of the radius of gyration tensor for neutral polymer chains are usually reported in terms of ratios, either between individual eigenvalues or with the mean square radius of gyration, since they are expected to attain universal values in the limit of long chains. For a chain with a spherically

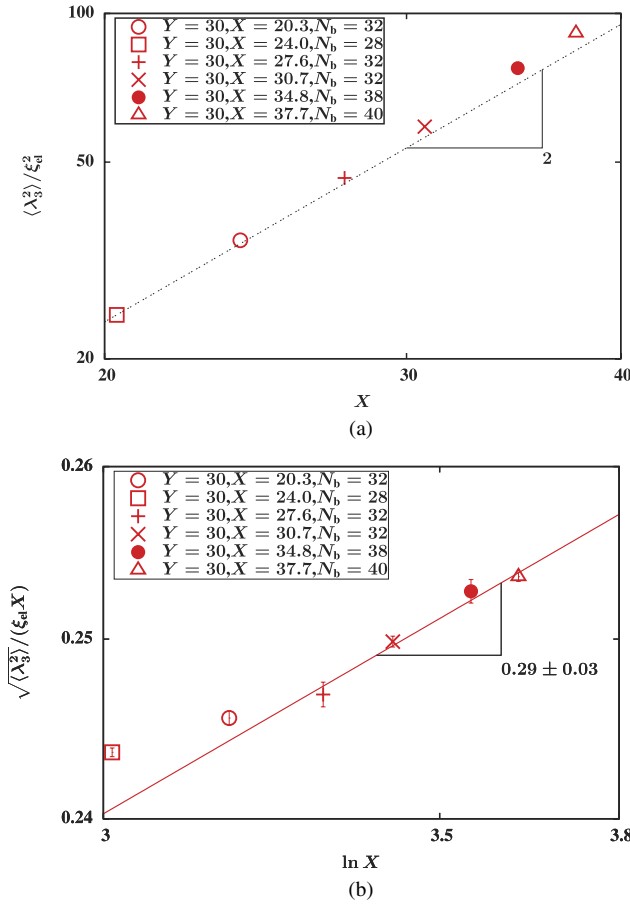


FIG. 7. Demonstration of the existence of logarithmic corrections to scaling at $Y = 30$ (in the blob-pole regime) for the largest scaled eigenvalue of the gyration tensor. (a) The dotted line shows that $\langle \lambda_3^2 \rangle / \xi_d^2$ departs from the X^2 scaling predicted by OSFKK theory for $X \gtrsim 30$. (b) A power law fit through the last four data points for $\sqrt{\langle \lambda_3^2 \rangle} / (\xi_d X)$ versus $\ln X$ (solid line) shows that the logarithmic corrections have an exponent close to the expected value of $1/3$. Data were obtained using the multi-chain algorithm with $c/c^* = 10^{-5}$. Error bars are of the order of symbol size in (a), and are explicitly displayed in (b).

symmetric shape about the centre of mass, we expect $\langle \lambda_i^2 \rangle / \langle r_g^2 \rangle = 1/3$, for $i = 1, 2, 3$, and $\langle \lambda_i^2 \rangle / \langle \lambda_j^2 \rangle = 1$ for all combinations i and j . Predictions reported previously in the literature for flexible neutral chains in θ -solutions, from a variety of different

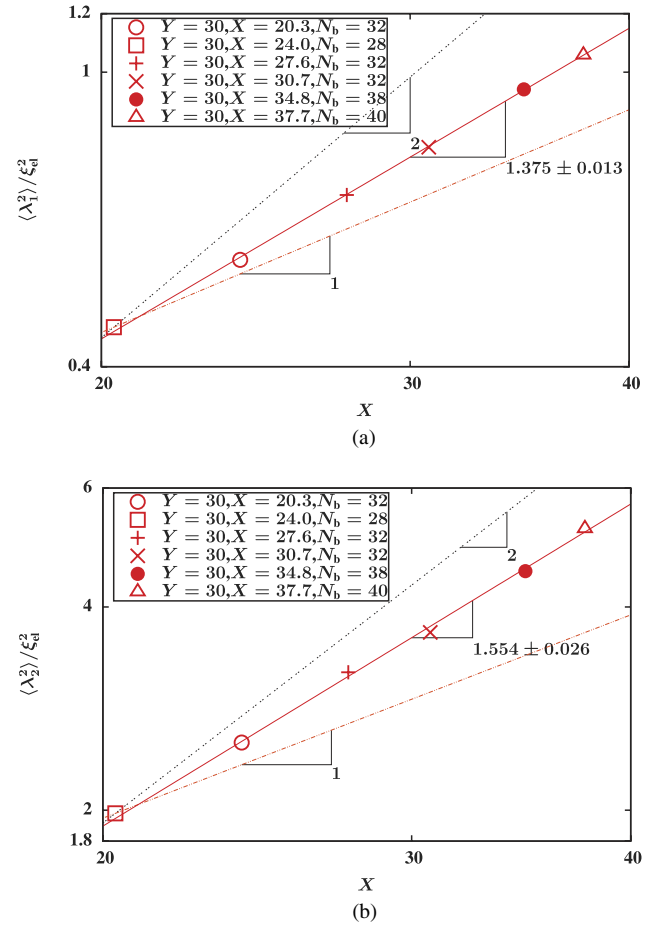


FIG. 8. Dependence on the number of blobs X , of (a) the smallest scaled eigenvalue of the gyration tensor, $\langle \lambda_1^2 \rangle / \xi_d^2$, and (b) the intermediate scaled eigenvalue of the gyration tensor, $\langle \lambda_2^2 \rangle / \xi_d^2$, at a reduced screening length, $Y = 30$. Data were obtained using the multi-chain algorithm with $c/c^* = 10^{-5}$. Error bars are of the order of symbol size.

approaches, are displayed in Table IV. The strong asymmetry in the shapes of chains is clearly apparent from the extent of departure from the expected values. Steinhäuser³⁰ has carried out molecular dynamics simulations to examine the influence of solvent quality on the ratios of eigenvalues, and noted that while their values are relatively similar in good and theta

TABLE IV. Normalized eigenvalues of the radius of gyration tensor for neutral random walk polymers compared with those for a polyelectrolyte chain (PE) in the ideal chain regime of the phase diagram. Predictions by Koyama¹⁵ and Wei²⁸ were obtained from approximate analytical calculations for infinite Gaussian chains, while those by Šolc,¹⁶ Šolc and Stockmayer¹⁷ and Kranbuehl and Verdier¹⁹ were obtained with lattice Monte Carlo simulations. Values reported from Theodorou and Suter²⁰ are for atactic polypropylene, obtained from Monte Carlo simulations based on a rotational isomeric state model, while those from Zifferer²⁷ are from lattice Monte Carlo simulations of “nonreversal random walks,” which are considered to represent θ -solvents. Data for the polyelectrolyte solution were obtained using the multi-chain algorithm, with $c/c^* = 10^{-5}$.

Solution	Chain length	$\langle \lambda_1^2 \rangle / \langle r_g^2 \rangle$	$\langle \lambda_2^2 \rangle / \langle r_g^2 \rangle$	$\langle \lambda_3^2 \rangle / \langle r_g^2 \rangle$	$\langle \lambda_2^2 \rangle / \langle \lambda_1^2 \rangle$	$\langle \lambda_3^2 \rangle / \langle \lambda_1^2 \rangle$
Neutral ¹⁵	$N_b \rightarrow \infty$	0.0646	0.175	0.754	2.5	10.6
Neutral ²⁸	$N_b \rightarrow \infty$	0.071	0.179	0.750	...	10.554
Neutral ^{16,17}	$N_b = 100$	0.065	0.175	0.75	2.7	11.7
Neutral ¹⁹	$N_b = 63$	0.065	0.176	0.76	2.70	11.7
Neutral ²⁰	$N_b = 999$	0.060 ± 0.004	0.17 ± 0.01	0.77 ± 0.06	2.9 ± 0.2	12.7 ± 0.9
Neutral ²⁷	$N_b \rightarrow \infty$	0.0633	0.1721	0.7645
PE ($Y = 0.01$)	$X = 24$	0.0631 ± 0.0003	0.1734 ± 0.0009	0.7626 ± 0.0061	2.75 ± 0.02	12.10 ± 0.12

solvents, they change significantly in poor solvents, with values that imply that chains become much more spherical as they collapse into globules. From OSFKK theory, we expect polyelectrolyte chains to behave like ideal chains for $Y \ll 1$. As can be seen from the last row of Table IV, this is indeed the case for the results obtained with the current simulations, with all the different ratios being in good agreement with results for neutral chains.

The dependence of the eigenvalues scaled with the mean square radius of gyration, on the reduced screening length Y , is displayed in Fig. 9. Starting with the limiting values reported in Table IV for polyelectrolyte chains in the ideal chain regime, both $\langle \lambda_1^2 \rangle / \langle r_g^2 \rangle$ and $\langle \lambda_2^2 \rangle / \langle r_g^2 \rangle$ (not shown here) decrease with increasing Y in the crossover regime until they reach their asymptotic values in the blob-pole regime. The reason for the decrease in the magnitude of these ratios is because $\langle r_g^2 \rangle = \langle \lambda_1^2 \rangle + \langle \lambda_2^2 \rangle + \langle \lambda_3^2 \rangle$ is dominated by the behaviour of $\langle \lambda_3^2 \rangle$, which can be seen from Fig. 6 to increase much more rapidly with Y than both $\langle \lambda_1^2 \rangle$ and $\langle \lambda_2^2 \rangle$. For the same reason, the ratio $\langle \lambda_3^2 \rangle / \langle r_g^2 \rangle$ increases with Y , nearly approaching a value of 1 in the limit of large Y . The overshoot in $\langle \lambda_1^2 \rangle$ and $\langle \lambda_2^2 \rangle$ that was observed in Figs. 6(a) and 6(b) at the threshold of the blob-pole regime is not noticeable when the eigenvalues are scaled with $\langle r_g^2 \rangle$ in place of ξ_{cl}^2 (as displayed in Fig. 9(a) for $\langle \lambda_1^2 \rangle / \langle r_g^2 \rangle$).

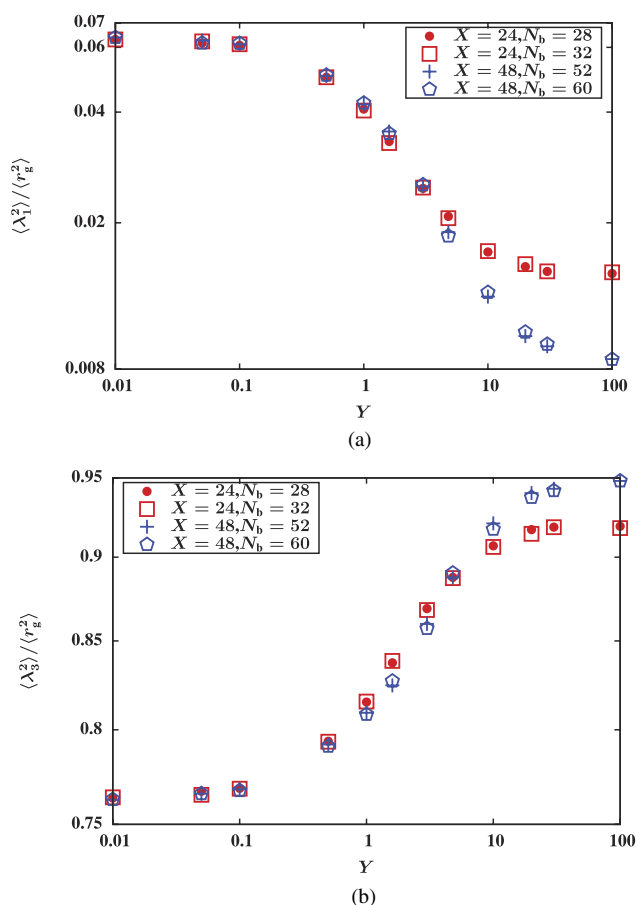


FIG. 9. Dependence of the eigenvalues of the radius of gyration tensor, normalised by the radius of gyration, on the reduced screening length Y , at two values of the number of blobs, X : (a) the smallest eigenvalue $\langle \lambda_1^2 \rangle / \langle r_g^2 \rangle$, and (b) the largest eigenvalue $\langle \lambda_3^2 \rangle / \langle r_g^2 \rangle$. Data were obtained using the multi-chain algorithm, with $c/c^* = 10^{-5}$. Error bars are of the order of symbol size.

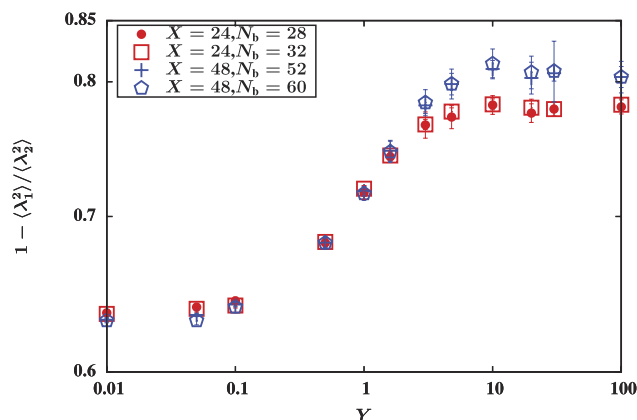


FIG. 10. Asymmetry in chain shape in the plane perpendicular to the stretching axis as a function of the reduced screening length, Y . Data were obtained using the multi-chain algorithm, with $c/c^* = 10^{-5}$.

The extent of asymmetry in chain shape in the plane perpendicular to the elongation axis can be gauged by the quantity, $1 - \langle \lambda_1^2 \rangle / \langle \lambda_2^2 \rangle$, which would be zero if the shape was symmetric. For neutral chains, and for polyelectrolyte chains in the ideal chain regime, using the values of $\langle \lambda_1^2 \rangle / \langle r_g^2 \rangle$ and $\langle \lambda_2^2 \rangle / \langle r_g^2 \rangle$ reported in Table IV, we can calculate $1 - \langle \lambda_1^2 \rangle / \langle \lambda_2^2 \rangle \approx 0.63$. Fig. 10 shows that with increasing electrostatic repulsion between blobs, the already highly asymmetric shape becomes even more so, levelling off to a constant value in the limit of large Y .

A notable aspect of the normalised eigenvalue ratios displayed in Figs. 9 and 10 is that their values are independent of the number of blobs X in the chain, for a wide variety of values of Y , ranging from the ideal chain regime and well into the crossover regime. The ratios become X dependent only as the chains approach the blob-pole regime with increasing screening length. This suggests that for sufficiently long chains, as in the case of neutral polymer chains, the normalised eigenvalues of polyelectrolyte chains attain universal values in those parts of the phase diagram that lie outside the blob-pole regime. Within the blob-pole regime, we expect that ratios of quantities that scale nearly identically with X will attain roughly constant values, such as $\langle \lambda_3^2 \rangle / \langle r_g^2 \rangle$ and $\langle \lambda_3^2 \rangle / \langle r_e^2 \rangle$ (which follows from the dominance of the growth of $\langle \lambda_3^2 \rangle$ with X , compared to $\langle \lambda_1^2 \rangle$ and $\langle \lambda_2^2 \rangle$), while ratios of quantities that do not scale identically would retain a dependence on the number of blobs. This is examined in Table V, where the dependence of a variety of ratios on the number of blobs X , at the particular value, $Y = 30$, which is in the blob-pole regime, is tabulated. At any fixed value of Y , however, we expect chains to move out of the blob-pole regime as $X \rightarrow \infty$, and for all the ratios to become universal.

C. Shape functions

For chain shapes with tetrahedral or greater symmetry, the asphericity $B = 0$, otherwise $B > 0$. For chain shapes with cylindrical symmetry, the acylindricity $C = 0$, otherwise $C > 0$. With regard to the degree of prolateness, its sign determines whether chain shapes are preponderantly oblate ($S, S^* \in [-0.25, 0]$) or prolate ($S, S^* \in [0, 2]$). The relative anisotropy

TABLE V. Normalized eigenvalues of the radius of gyration tensor for various values of the number of blobs X , at a reduced screening length, $Y = 30$. Data were obtained using the multi-chain algorithm, with $c/c^* = 10^{-5}$.

X	$\langle\lambda_1^2\rangle/\langle r_g^2\rangle$	$\langle\lambda_2^2\rangle/\langle r_g^2\rangle$	$\langle\lambda_3^2\rangle/\langle r_g^2\rangle$	$\langle\lambda_2^2\rangle/\langle\lambda_1^2\rangle$	$\langle\lambda_3^2\rangle/\langle\lambda_1^2\rangle$	$1 - \langle\lambda_1^2\rangle/\langle\lambda_2^2\rangle$
20.3	0.0168 ± 0.0001	0.0734 ± 0.0003	0.9099 ± 0.0026	4.37 ± 0.02	54.2 ± 0.22	0.771 ± 0.001
24	0.0148 ± 0.0001	0.0666 ± 0.0004	0.9186 ± 0.0038	4.52 ± 0.03	62.3 ± 0.30	0.779 ± 0.002
27.6	0.0136 ± 0.0001	0.0636 ± 0.0011	0.9228 ± 0.0078	4.69 ± 0.09	68.1 ± 0.88	0.787 ± 0.004
30.7	0.0125 ± 0.0001	0.0578 ± 0.0004	0.9297 ± 0.0034	4.63 ± 0.04	74.5 ± 0.43	0.784 ± 0.002
34.8	0.0114 ± 0.0001	0.0545 ± 0.0011	0.9341 ± 0.0076	4.77 ± 0.11	81.8 ± 1.12	0.790 ± 0.004
37.7	0.0108 ± 0.0001	0.0536 ± 0.0004	0.9356 ± 0.0030	4.95 ± 0.04	86.5 ± 0.40	0.798 ± 0.001
48	0.0094 ± 0.0002	0.0488 ± 0.0013	0.9418 ± 0.0074	5.22 ± 0.16	100.7 ± 1.7	0.808 ± 0.006

(κ^2 and κ^{2*}), on the other hand, lies between 0 (for spheres) and 1 (for rods). In this context, the values of the normalised shape functions $B/\langle r_g^2\rangle$ and $C/\langle r_g^2\rangle$, S and S^* , and κ^2 and κ^{2*} , predicted by various techniques for neutral polymer chains are compared with the predictions of the current simulations for polyelectrolyte chains in the ideal chain regime, in Table VI. Clearly, in this regime, polyelectrolyte chains behave identically to neutral random walk chains, and share the same high degree of anisotropy.

The dependence of the shape functions $B/\langle r_g^2\rangle$, $C/\langle r_g^2\rangle$, S^* and κ^{2*} on Y , for two different values of X , is displayed in Fig. 11. As we might expect, chains appear to become more aspherical, more cylindrical, more prolate, and more rodlike with increasing electrostatic repulsion between the blobs. The functions S and κ^2 behave similarly to S^* and κ^{2*} .

The independence of the values of the shape functions from the number of blobs X , in all the regimes except the blob-pole regime, suggests that for sufficiently long chains, at any given value of the reduced screening length Y , the shapes of polyelectrolyte chains are universal. In the blob-pole regime however, as noted earlier, the appearance of corrections to scaling (to varying extents in the three principal directions) leads to a dependence of chain shapes on the chain length. The nature of this dependence on X , at a particular value of $Y = 30$, is displayed in Fig. 12 for B/ξ_{el}^2 , C/ξ_{el}^2 , S , and κ^2 . The functions S^* and κ^{2*} behave similarly to S and κ^2 .

D. Translational diffusivity interpreted with the OSFKK scaling picture

The dependence on the reduced screening length Y , of all the static properties examined here so far, has followed a

similar pattern: they exhibit Y independent behaviour in the ideal chain and blob-pole regimes, while they are functions of Y in the crossover regime between these two limits. By defining the Zimm diffusivity of an electrostatic blob through the expression

$$D_\xi = \frac{k_B T}{6\pi\eta_s(\xi_{el}/2)}, \quad (33)$$

we find that a similar pattern is obeyed by the ratio D/D_ξ , where D is the translational diffusivity of the polyelectrolyte chain as a whole.

In the ideal chain regime, since $D \sim k_B T/(6\pi\eta_s r_e)$, we expect $D/D_\xi \sim X^{-1/2}$. In the blob-pole regime, an expression for the diffusivity can be derived by drawing an analogy with the shish-kebab model for rodlike polymers. In the latter, a rodlike polymer of length L is modelled as $N = (L/d)$ beads of diameter d , placed along a straight line. The translational diffusivity of such a shish-kebab can be shown to be³³

$$D = \frac{\ln(L/d)}{3\pi\eta_s L} k_B T. \quad (34)$$

By mapping $d \rightarrow \xi_{el}$, and $L \rightarrow r_e$, it follows that the diffusivity of a blob-pole is given by

$$D = \frac{\ln(X)}{3\pi\eta_s r_e} k_B T. \quad (35)$$

Within the OSFKK scaling ansatz, since $(r_e/\xi_{el}) \sim X$ in the blob-pole regime, we see that

$$\frac{D}{D_\xi} = \frac{\ln(X)}{X}. \quad (36)$$

If logarithmic corrections to the scaling of r_e (according to Eq. (5)) are taken into account, we expect

TABLE VI. Shape functions for neutral random walk polymers compared with those for a polyelectrolyte chain (PE) in the ideal chain regime of the phase diagram. Predictions by Steinhäuser³⁰ were obtained with molecular dynamics simulations, while those by Bishop and Michels²² and Bishop *et al.*²⁶ were obtained with Brownian dynamics, and off-lattice Monte Carlo simulations, respectively. Methods used to evaluate shape functions in Refs. 20, 27, and 28 are given in the caption to Table IV. Data for the polyelectrolyte solution were obtained using the multi-chain algorithm, with $c/c^* = 10^{-5}$.

Solution	Chain length	$B/\langle r_g^2\rangle$	$C/\langle r_g^2\rangle$	S^*	S	κ^{2*}	κ^2
Neutral ²⁸	$N_b \rightarrow \infty$	0.4750	0.8875	0.3943	0.5263
Neutral ²⁰	$N_b = 999$	0.66 ± 0.04	0.106 ± 0.007	0.41 ± 0.01	...
Neutral ²²	$N_b = 48$	0.460 ± 0.021	0.859 ± 0.245	0.384 ± 0.011	0.518 ± 0.095
Neutral ²⁶	$N_b \rightarrow \infty$	0.397 ± 0.001	0.529 ± 0.001
Neutral ³⁰	$N_b \rightarrow \infty$	0.6255 ± 0.0005	0.394 ± 0.003	0.5221 ± 0.0013
Neutral ²⁷	$N_b \rightarrow \infty$	0.4751	0.8875	0.3943	0.5264
PE ($Y = 0.01$)	$X = 24$	0.6453 ± 0.0056	0.1103 ± 0.0006	0.4751 ± 0.0024	0.8810 ± 0.0205	0.3955 ± 0.0011	0.5253 ± 0.0074

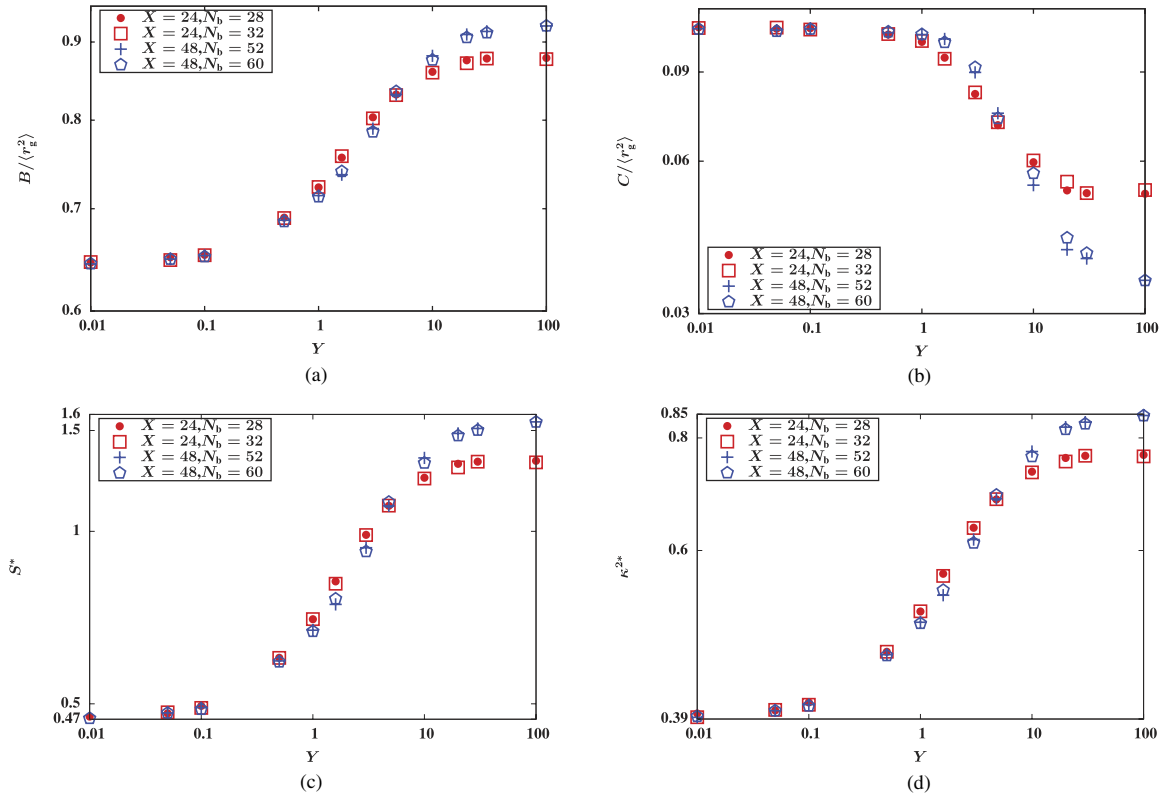


FIG. 11. The dependence of shape functions on the reduced screen length Y , at two values of X : (a) the normalised asphericity $B/\langle r_g^2 \rangle$ (see Eq. (19)), (b) the normalised acylindricity $C/\langle r_g^2 \rangle$ (see Eq. (20)), (c) the degree of prolateness, S^* (see Eq. (21)) and (d) the shape anisotropy, κ^{2*} (see Eq. (23)). When not displayed explicitly, error bars are of the order of symbol size. Data obtained with the multi-chain code, with $c/c^* = 10^{-5}$.

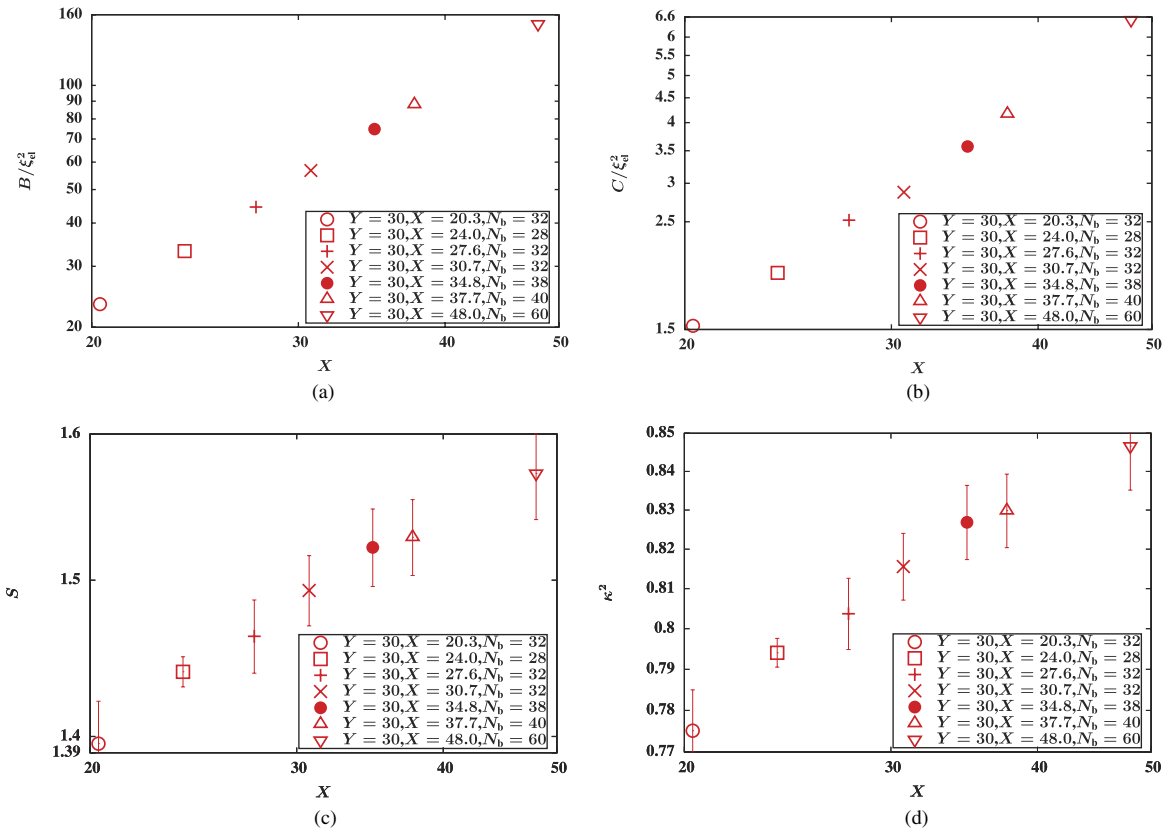


FIG. 12. The dependence of shape functions on the number of blobs X , in the blob-pole regime, at a particular value of the reduced screening length $Y = 30$: (a) the scaled asphericity B/ξ_{el}^2 , (b) the scaled acylindricity C/ξ_{el}^2 , (c) the degree of prolateness S , and, (d) the shape anisotropy κ^2 . Error bars are of the order of symbol size in (a) and (b). Data obtained with the multi-chain code, with $c/c^* = 10^{-5}$.

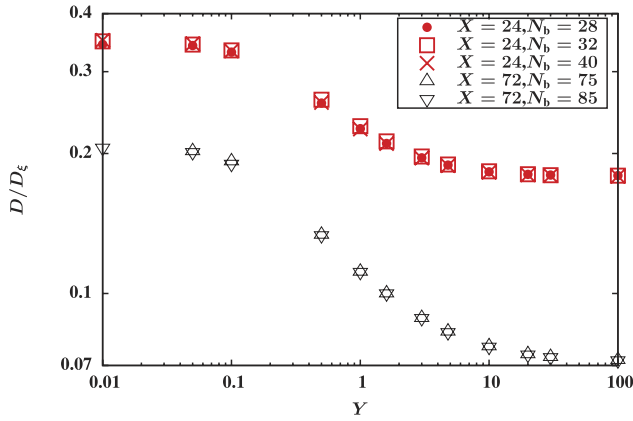


FIG. 13. Dependence of the ratio of the translational diffusivity of the chain D , to the diffusivity of the electrostatic blob D_ξ , on the reduced screening length Y , at two values of the number of blobs, X . Error bars are of the order of symbol size.

$$\frac{D}{D_\xi} = \frac{[\ln(X)]^{2/3}}{X}. \quad (37)$$

In either case, we see that the ratio D/D_ξ is independent of Y in the both the limiting regimes of small and large Y . In between these two limits, we expect D/D_ξ to depend on Y .

These arguments are substantiated in the plot of D/D_ξ versus Y , at two values of X , displayed in Fig. 13. We see that for each value of X , the ratio D/D_ξ crosses over from a constant value in the ideal chain regime to a constant value in the blob-pole regime. As has been observed in the case of all the properties examined so far, the use of blob scaling variables to interpret the results of BD simulations leads to behaviour that is independent of the specific choice of bead-spring chain parameters.

The importance of accounting for logarithmic corrections to the scaling of r_e with X in Eq. (35), is studied in Fig. 14. The plot of $D/(D_\xi \ln X)$ versus $1/X$ in Fig. 14(a) shows that while the scaling described by Eq. (36) is obeyed at relatively small values of X , there is a departure from OSFCK scaling for $X \gtrsim 30$. On the other hand, the plot of $D/(D_\xi X^{-1})$ as a function of $\ln X$ displayed in Fig. 14(b) shows that Eq. (37) is indeed obeyed, with the exponent of $\ln X$ close to the expected value of $2/3$.

A comparison of the magnitude of the diffusivity in the ideal chain and blob-pole regimes in Fig. 13 suggests that strongly screened chains ($Y \rightarrow 0$) diffuse more than twice as fast as nearly unscreened chains ($Y \gg 1$), and that this ratio increases with X . This behaviour is examined in more detail in Fig. 15, which displays the ratio of the diffusivity of a chain with $Y = 0.01$ to that of a chain with $Y = 30$, for various values of X . When logarithmic scaling corrections are ignored, we expect this ratio to scale as $X^{1/2}$. As can be seen from Fig. 15, this scaling is approximately satisfied. As in other examples considered here, the inclusion of logarithmic corrections in the blob-pole regime leads to a better fit of the data, as shown in the inset of Fig. 15.

From the scaling of r_e/ξ_{el} (see Eq. (5)) and D/D_ξ (see Eq. (37)) with X in the blob-pole regime, it can be seen that the ratio

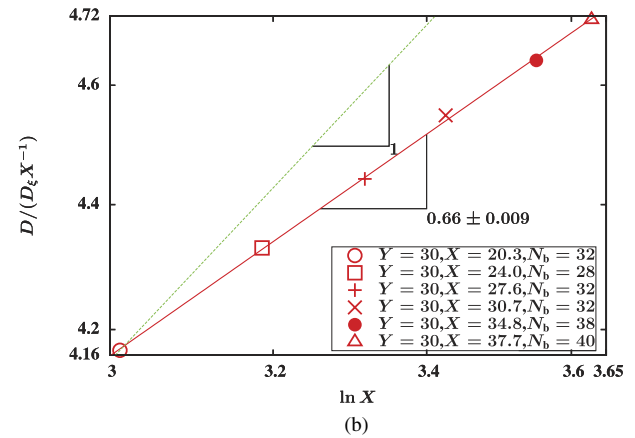
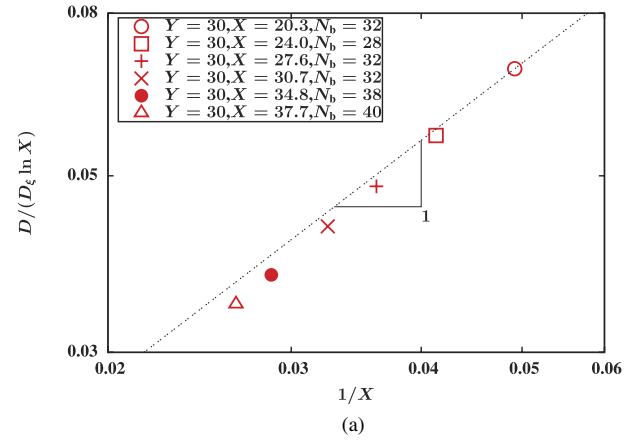


FIG. 14. Demonstration of the importance of accounting for logarithmic corrections to the scaling of r_e with X in Eq. (35) for the translational diffusivity D in the blob-pole regime: (a) departure from OSFCK scaling at large values of X (see Eq. (36)). (b) Scaling behaviour in agreement with Eq. (37), with the expected exponent of $\ln X \approx 2/3$. Error bars are of the order of symbol size.

$$\frac{\langle r_e^2 \rangle / \xi_{el}^2}{D/D_\xi} = \frac{\langle r_e^2 \rangle / D}{\tau_\xi} \sim X^3,$$

where τ_ξ , the blob relaxation time, is defined by $\tau_\xi = \xi_{el}^2/D_\xi$. Notably, this ratio is free from logarithmic corrections even in the blob-pole regime. Figure 16(a) displays the dependence on

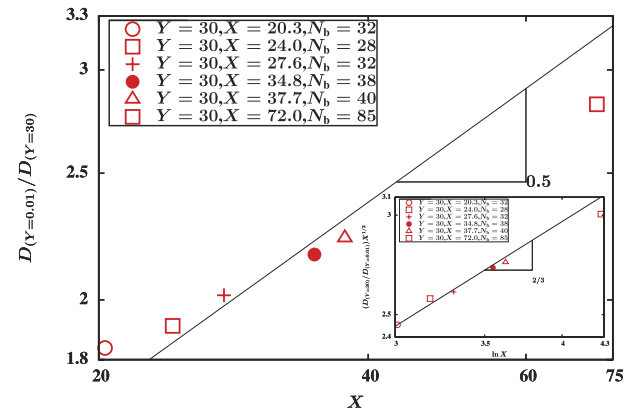


FIG. 15. Dependence on the number of blobs, X , of the ratio of the translational diffusivity of the chain in the ideal chain regime ($Y = 0.01$) to the diffusivity of a chain in the blob-pole regime ($Y = 30$). Inset shows the importance of accounting for logarithmic corrections to scaling. Error bars are of the order of symbol size.

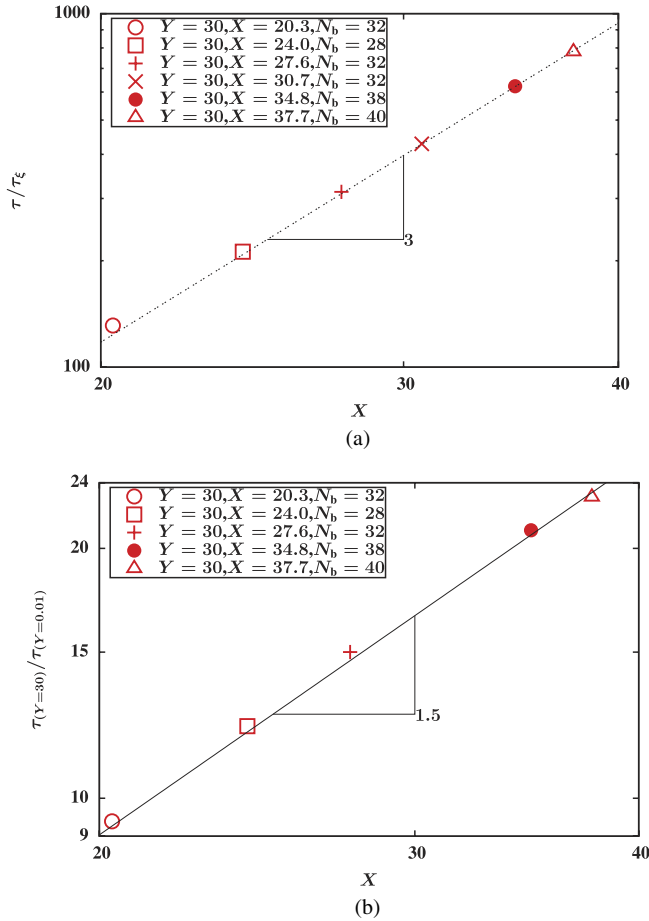


FIG. 16. Dependence on the number of blobs, X , of (a) the ratio of the relaxation time of the chain in the blob-pole regime to the relaxation time of an electrostatic blob, and (b) the ratio of the relaxation time of the chain in the blob-pole regime (at $Y = 30$) to the relaxation time in the ideal chain regime (at $Y = 0.01$). Error bars are of the order of symbol size.

the number of blobs X of the ratio of the relaxation time in the blob-pole regime to the relaxation time of an electrostatic blob, τ/τ_ξ . Though the relaxation time τ has been defined in terms of $\langle r_g^2 \rangle$ instead of $\langle r_e^2 \rangle$ (see Eq. (26)), we see from Fig. 16(a) that (in the range of X values that have been considered) any residual logarithmic corrections are extremely weak, and that the ratio scales with X with the expected power law exponent.

The difference in the conformations of the chain in the two regimes leads to the long-time relaxation in the blob-pole regime being nearly an order of magnitude slower than in the ideal chain regime, as displayed in Fig. 16(b). As can be seen from the figure, the ratio of the two limiting relaxation times, $\tau(Y=30)/\tau(Y=0.01)$, increases with the number of blobs X with the expected power law, $X^{3/2}$.

The ratio U_{RD} , defined by

$$U_{RD} = \frac{r_g}{r_h}, \quad (38)$$

where r_h is the hydrodynamic radius, $r_h = k_B T / (6\pi\eta_s D)$, has an universal value for neutral polymer solutions in the long chain limit, since both r_g and r_h scale identically with chain length.^{34,42,50} In particular, by extrapolating finite chain data

acquired from highly accurate BD simulations to the long chain limit, Sunthar and Prakash⁴⁴ have shown that for θ -solutions, $U_{RD}^\theta = 1.38 \pm 0.01$. We anticipate that for the polyelectrolyte solutions considered here, U_{RD} will have a similar value in the ideal chain regime.

In the blob-pole regime, by substituting for D from Eq. (35) into Eq. (38), we see that

$$U_{RD} \sim \left(\frac{r_g}{r_e} \right) \ln X. \quad (39)$$

As a result, U_{RD} does not have a universal value in this regime, but rather depends on the number of blobs X . In the context of the current simulations, Figs. 3(b) and 4(b) show that the scaling expressions, $r_e/(\xi_{el}X) \sim (\ln X)^{0.36}$ and $r_g/(\xi_{el}X) \sim (\ln X)^{0.22}$, respectively, provide good fits to the simulation data. As a result, we expect from Eq. (39) that $U_{RD} \sim (\ln X)^{0.86}$.

Figure 17(a) displays the dependence of U_{RD} on the reduced screening length Y , for two different values of X . We see that for small values of Y , the ratio has a constant universal value which is close to that for neutral chains. The ratio increases, while remaining universal (i.e., independent of X) for increasing values of Y in the crossover regime, before levelling off to a non-universal value in the blob-pole regime.

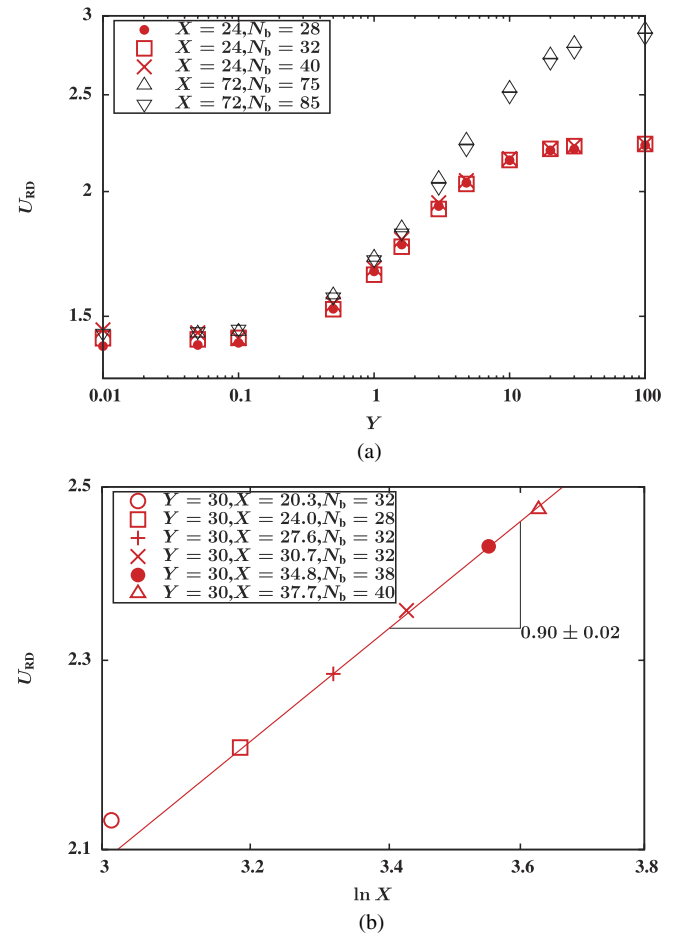


FIG. 17. Ratio of the radius of gyration to the hydrodynamic radius, U_{RD} (see Eq. (38)), as a function of (a) the reduced screening length Y , at two different values of the number of blobs X , and (b) demonstration of logarithmic corrections to scaling in the blob-pole regime, at a constant value of the reduced screening length, $Y = 30$. Error bars are of the order of symbol size.

The dependence of U_{RD} on $\ln X$ in the blob-pole regime, at the particular value $Y = 30$, is displayed in Fig. 17(b). We see that the exponent of $\ln X$ is indeed close to the expected value of 0.86.

IV. CONCLUSIONS

The size, shape, and diffusivity of a weakly charged polyelectrolyte chain in solution have been examined in the limit of low polymer concentration using Brownian dynamics simulations of a coarse-grained bead-spring chain model, with Debye–Hückel electrostatic interactions between the beads. Simulation results have been recast in terms of the scaling variables X (the number of electrostatic blobs), and Y (the reduced screening length), which are defined within the framework of the OSFKK blob scaling theory. While the root-mean-square end-to-end vector and radius of gyration have been used to represent the mean size of the chain, various functions defined in terms of the eigenvalues of the radius of gyration tensor have been used to characterise chain shape in all parts of the $\{X, Y\}$ phase-space. The translational diffusion coefficient of the chain, which is a dynamic property, has been determined accurately from the displacement of the chain centre of mass, by taking hydrodynamic interactions into account through incorporation of the Rotne-Prager-Yamakawa (RPY) tensor into the BD algorithm.

The key results of the present work are summarised in the list below.

1. Interpretation of simulation results in terms of blob scaling variables X and Y leads to a description of solution behaviour independent of the level of coarse-graining, i.e., of the choice of the number of beads N_b , and other parameters in the bead-spring chain model. This should prove extremely useful for comparing simulation results with experiments. The procedure for mapping from experimental variables to blob scaling variables, in order to facilitate this comparison, has been provided in Appendix B.
2. Three broad domains of behaviour can be clearly identified: (i) the ideal chain regime corresponding to small values of Y , (ii) the crossover regime corresponding to intermediate values of Y , and (iii) the blob-pole regime corresponding to large values of Y .
3. In the blob-pole regime, BD simulations appear to pick up the existence of logarithmic corrections for fairly short chains, which are in good agreement with predictions of refined scaling theories.
4. Various universal ratios of eigenvalues, the asphericity, the acylindricity, the degree of prolateness, and the shape anisotropy of polyelectrolyte chains in the ideal chain regime have been compared with published results in the literature for flexible neutral random walk polymers.
5. When suitably scaled, the size, shape, and diffusivity of a chain are independent of the reduced screening length Y in the ideal chain and blob-pole regimes, and depend on Y only in the crossover regime.
6. The translational diffusivity of the chain in the blob-pole regime can be described by drawing an analogy with the translational diffusivity of a rodlike polymer modelled as a

shish-kebab. The normalisation parameter that enables the collapse of BD data for the translational diffusivity is the Zimm diffusivity of the electrostatic blob.

7. All properties of polyelectrolyte solutions, when suitably normalised, appear to exhibit universal behaviour, independent of the number of electrostatic blobs X in the chain, in all regimes of the $\{X, Y\}$ phase-space, except in the blob-pole regime, where the occurrence of logarithmic corrections to scaling leads to non-universal behaviour.

ACKNOWLEDGMENTS

The authors gratefully acknowledge CPU time grants from the National Computational Infrastructure (NCI) facility hosted by the Australian National University, and Victorian Life Sciences Computation Initiative (VLSCI) hosted by the University of Melbourne.

APPENDIX A: SCALING OF THE END-TO-END VECTOR IN THE BLOB-POLE REGIME

The scaling of the end-to-end vector in the blob-pole regime can be derived from a Flory type energy minimisation argument. The essential assumption is that chains adopt an ellipsoidal shape, with electrostatic interactions causing stretching in the direction of the major-axis, while leaving chain conformations in the direction of the minor-axes unaffected. As a result, the long half-axis of the ellipsoid scales as $r_e/2$, while the aspect ratio scales as $r_e/(b_K N_K^{1/2})$. The total energy of such a chain can be obtained by combining the electrostatic energy of an ellipsoid with the elastic energy of a stretched chain. Ignoring prefactors of order unity, the electrostatic energy of an ellipsoid is¹⁰

$$k_B T \frac{l_B (f N_K)^2}{r_e} \ln \left(\frac{r_e}{b_K N_K^{1/2}} \right)$$

while the elastic energy of a stretched chain is¹

$$k_B T \left(\frac{r_e^2}{b_K^2 N_K} \right).$$

From Flory theory, it follows that the equilibrium end-to-end vector can be derived by minimising the total energy,

$$\frac{U}{k_B T} \sim \left(\frac{r_e^2}{b_K^2 N_K} \right) + \frac{l_B (f N_K)^2}{r_e} \ln \left(\frac{r_e}{b_K N_K^{1/2}} \right). \quad (A1)$$

It is convenient to proceed in two steps—first deriving the equilibrium end-to-end vector by neglecting the logarithmic term in the electrostatic energy, and then accounting for it in the next step. Minimising the sum,

$$\left(\frac{r_e^2}{b_K^2 N_K} \right) + \frac{l_B (f N_K)^2}{r_e},$$

with respect to r_e leads to

$$r_e = f^{2/3} l_B^{1/3} N_K b_K. \quad (A2)$$

This is identical to the expression derived by OSFKK theory for the end-to-end vector in the blob-pole regime (see Table I). We now assume that including the logarithmic term in the electrostatic energy leads to a modification of the equilibrium end-to-end vector,

$$r_e = f^{\frac{2}{3}} \hat{l}_B^{\frac{1}{3}} N_K b_K x, \quad (\text{A3})$$

where the factor x remains to be determined. Substituting the modified expression for r_e from Eq. (A3) into Eq. (A1) (and absorbing the common factor $N_K(f^2 \hat{l}_B)^{\frac{2}{3}}$ into the energy), leads to the following sum:

$$x^2 + \frac{1}{x} \ln \left[N_K^{\frac{1}{2}} f^{\frac{2}{3}} \hat{l}_B^{\frac{1}{3}} x \right]$$

which must be minimised with respect to x in order to determine x . This leads to

$$x^3 = \ln \left[N_K^{\frac{1}{2}} f^{\frac{2}{3}} \hat{l}_B^{\frac{1}{3}} \right] + \ln x - 1. \quad (\text{A4})$$

Assuming $x \gg 1$ implies

$$x = \left\{ \frac{1}{2} \ln \left[N_K f^{\frac{4}{3}} \hat{l}_B^{\frac{2}{3}} \right] \right\}^{\frac{1}{3}}, \quad (\text{A5})$$

where the factor $(1/2)$ in front of the logarithmic term has been introduced for future convenience. The assumption that x is large can be seen to be justified for $N_K \gg 1$.

Substituting the expression for x from Eq. (A5) into Eq. (A3) shows that this two-step procedure leads to a prediction of the equilibrium end-to-end vector scaling with chain size according to

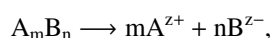
$$\frac{r_e}{b_K} \sim f^{\frac{2}{3}} \hat{l}_B^{\frac{1}{3}} N_K \left\{ \ln \left[N_K f^{\frac{4}{3}} \hat{l}_B^{\frac{2}{3}} \right] \right\}^{\frac{1}{3}}. \quad (\text{A6})$$

Clearly, the logarithmic correction to the OSFKK scaling expression in the blob-pole regime comes from taking the logarithmic term in the expression for the electrostatic energy of an ellipsoid into account. Equation (A6) is similar to the expression derived previously by Dobrynin and Rubinstein.¹⁰ This expression takes a particularly simple form in terms of blob scaling variables (see Eqs. (4) and (2)),

$$\frac{r_e}{\xi_{el}} \sim X [\ln X]^{\frac{1}{3}}. \quad (\text{A7})$$

APPENDIX B: MAPPING EXPERIMENTS TO BLOB VARIABLES FOR NaPSS

Consider a polyelectrolyte chain of molecular weight M , dissolved in a solvent at temperature T , with a temperature dependent relative permittivity $\epsilon_r(T)$. If the monomer concentration in molar units is c_p^m , and the degree of ionization per chain is α , then the number of counterions in solution per unit volume is, $\alpha c_p^m(N_A 1000)$, where N_A is Avagadro's number. If the added salt $A_m B_n$, with molar concentration c_s^m , dissociates in solution according to



where z_+ and z_- are the cation and anion valences, respectively,

then the number of cations A per unit volume in solution is $c_+ = m c_s^m(N_A 1000)$, and the number of anions B per unit volume is $c_- = n c_s^m(N_A 1000)$. It follows that the ionic strength I of the solution is given by

$$I = \frac{1}{2} (m z_+^2 + n z_-^2) c_s^m(N_A 1000) + \frac{1}{2} z_p^2 \alpha c_p^m(N_A 1000), \quad (\text{B1})$$

where z_p is the counterion valence.

The Bjerrum length l_B can be calculated from Eq. (1), while the Debye length is given by

$$l_D = (8\pi l_B I)^{-\frac{1}{2}}. \quad (\text{B2})$$

Mapping the experimental system onto blob variables is straightforward once it is mapped onto the bare-model parameters $\{N_K, f, \hat{l}_B, \hat{l}_D\}$. In order to do so, it is necessary to know the intrinsic persistence length of the polyelectrolyte chain, ℓ_0 , the molecular weight of a monomer, M_m , and the monomer length b_m . With this information, we can find (i) the Kuhn step length, $b_K = 2\ell_0$, (ii) the number of monomers in a chain, $N_m = M/M_m$, (iii) the number of Kuhn steps, $N_K = (N_m b_m)/b_K$, and (iv) the number of charges per Kuhn step, $f = \alpha N_{m,K}$, where $N_{m,K} = b_K/b_m$, is the number of monomers per Kuhn step. It is also straight forward to find $\hat{l}_B = l_B/b_K$ and $\hat{l}_D = l_D/b_K$ at the given temperature and ionic strength. Once the bare-model parameters are known, the blob scaling variables, $\{X, Y, \xi_{el}\}$ can be determined from Eqs. (2)–(4).

Table VII displays the blob scaling variables calculated with this procedure for sodium poly(styrene sulfonate) (NaPSS) at two different molecular weights, $M = 20\,600$ Dalton and $M = 206\,000$ Dalton. For NaPSS, the molecular weight of a monomer, $M_m = 206$ Dalton, and the monomer length $b_m = 2.5$ Å.⁵¹ The intrinsic persistence length of NaPSS is variously estimated^{51–53} as lying in the range $9 \text{ Å} \leq \ell_0 \leq 14 \text{ Å}$. For the purposes of this illustration, we choose a value $\ell_0 = 12.5$ Å in order to get an integer number of monomers per Kuhn step. Similarly, a fixed value of $c_p^m = 10^{-6}$ mol/l is chosen as the monomer concentration, since at this concentration, according to the phase diagram in Ref. 52, the solution remains in the dilute regime for all the values of N_m considered here. The degree of ionization (sulfonation) per chain for NaPSS is commonly assumed to be $\alpha = 0.35$.^{51–53} Here, we choose a range of values between 0.01 and 0.35 in order to generate a range of values of the blob scaling variables X and Y . For water,⁵⁴ $\epsilon_r(t) = 87.740 - 0.40008 t + 9.398 \times 10^{-4} t^2 - 1.410 \times 10^{-6} t^3$, where $0^\circ \text{C} \leq t \leq 100^\circ \text{C}$. We assume monovalent cations and anions for the added salt, with composition stoichiometry $m = 1$ and $n = 1$. The salt concentration c_s^m is varied in order to change the ionic strength, and consequently, the Debye length. Finally, three values of temperature are chosen that bracket typical room temperature values.

As mentioned earlier, in the current simulation, blob variable values in the range $20 < X < 80$ and $0.01 \leq Y \leq 100$ have been considered. It is clear from Table VII that by a suitable choice of experimental parameters, it is possible to span this range of values of X and Y , while maintaining the Manning parameter, $\gamma_0 \lesssim 1$.

TABLE VII. Mapping experimental data for sodium poly(styrene sulfonate) to the blob model.

$M = 20\,600$ ($N_{\text{m}} = 100, N_{\text{K}} = 10$)									
Solution properties						Blob variables			
α (f)	c_{s}^{m} (I) (mol/l)	T °C	l_{B} (nm)	l_{D} (nm)	γ_0	ξ_{el} (nm)	X	Y	
0.01 (0.1)	$0\,(5 \times 10^{-9})$	15	0.7077	4321.0	0.0283	17.67	0.2001	244.5084	
		25	0.7158	4297.0	0.0286	17.61	0.2016	244.0463	
		35	0.7247	4270.0	0.0290	17.53	0.2033	243.5422	
	0.01 (0.01)	15	0.7077	3.056	0.0283	17.67	0.2001	0.1729	
		25	0.7158	3.038	0.0286	17.61	0.2016	0.1726	
		35	0.7247	3.019	0.0290	17.53	0.2033	0.1722	
	0.5 (0.5)	15	0.7077	0.4321	0.0283	17.67	0.2001	0.0245	
		25	0.7158	0.4297	0.0286	17.61	0.2016	0.0244	
		35	0.7247	0.4270	0.0290	17.53	0.2033	0.0244	
	$0\,(1.5 \times 10^{-7})$	15	0.7077	788.9	0.8492	1.830	18.6534	431.0033	
		25	0.7158	784.5	0.8589	1.824	18.7951	430.1889	
		35	0.7247	779.6	0.8696	1.816	18.9512	429.3002	
0.3 (3.0)	0.01 (0.01)	15	0.7077	3.056	0.8492	1.830	18.6534	1.6693	
		25	0.7158	3.038	0.8589	1.824	18.7951	1.6661	
		35	0.7247	3.019	0.8696	1.816	18.9512	1.6627	
	0.5 (0.5)	15	0.7077	0.4321	0.8492	1.830	18.6534	0.2361	
		25	0.7158	0.4297	0.8589	1.824	18.7951	0.2356	
		35	0.7247	0.4270	0.8696	1.816	18.9512	0.2351	
$M = 206\,000$ ($N_{\text{m}} = 1000, N_{\text{K}} = 100$)									
0.1 (1.0)	4×10^{-6} (4.05×10^{-6})	15	0.7077	151.8	0.2831	3.808	43.1119	39.8766	
		25	0.7158	151.0	0.2863	3.793	43.4393	39.8012	
		35	0.7247	150.0	0.2899	3.777	43.8001	39.7190	
	0.1 (0.1)	15	0.7077	0.9662	0.2831	3.808	43.1119	0.2538	
		25	0.7158	0.9608	0.2863	3.793	43.4393	0.2533	
		35	0.7247	0.9548	0.2899	3.777	43.8001	0.2528	
	0.8 (0.8)	15	0.7077	0.3416	0.2831	3.808	43.1119	0.0897	
		25	0.7158	0.3397	0.2863	3.793	43.4393	0.0896	
		35	0.7247	0.3376	0.2899	3.777	43.8001	0.0894	
	0.35 (3.5)	4×10^{-6} (4.175×10^{-6})	15	0.7077	149.5	0.9907	1.652	229.0977	90.5376
			25	0.7158	148.7	1.0021	1.645	230.8376	90.3666
			35	0.7247	147.8	1.0146	1.639	232.7550	90.1799
0.1 (0.1)		15	0.7077	0.9662	0.9907	1.652	229.0977	0.5850	
		25	0.7158	0.9608	1.0021	1.645	230.8376	0.5839	
		35	0.7247	0.9548	1.0146	1.639	232.7550	0.5827	
0.8 (0.8)		15	0.7077	0.3416	0.9907	1.652	229.0977	0.2068	
		25	0.7158	0.3397	1.0021	1.645	230.8376	0.2064	
		35	0.7247	0.3376	1.0146	1.639	232.7550	0.2060	

¹P.-G. de Gennes, *Scaling Concepts in Polymer Physics* (Cornell University Press, 1979).²M. Rubinstein and R. H. Colby, *Polymer Physics* (Oxford University Press, 2003).³P.-G. de Gennes, P. Pincus, R. M. Velasco, and F. Brochard, *J. Phys. France* **37**, 1461 (1976).⁴T. Odijk, *J. Polym. Sci., Polym. Phys. Ed.* **15**, 477 (1977).⁵J. Skolnick and M. Fixman, *Macromolecules* **10**, 944 (1977).⁶A. Khokhlov and K. Khachaturian, *Polymer* **23**, 1742 (1982).⁷R. Everaers, A. Milchev, and V. Yamakov, *Eur. Phys. J. E* **8**, 3 (2002).⁸J. Rotne and S. Prager, *J. Chem. Phys.* **50**, 4831 (1969).⁹H. Yamakawa, *J. Chem. Phys.* **53**, 436 (1970).¹⁰A. V. Dobrynin and M. Rubinstein, *Prog. Polym. Sci.* **30**, 1049 (2005).¹¹S. K. Pattanayek and J. R. Prakash, *Macromolecules* **41**, 2260 (2008).¹²C. Stoltz, J. J. de Pablo, and M. D. Graham, *J. Chem. Phys.* **126**, 124906 (2007).¹³G. S. Manning, *J. Chem. Phys.* **51**, 924 (1969).¹⁴W. Kuhn, *Kolloid-Z.* **68**, 2 (1934).¹⁵R. Koyama, *J. Phys. Soc. Jpn.* **24**, 580 (1968).¹⁶K. Šolc, *J. Chem. Phys.* **55**, 335 (1971).¹⁷K. Šolc and W. H. Stockmayer, *J. Chem. Phys.* **54**, 2756 (1971).¹⁸D. Y. Yoon and P. J. Flory, *J. Chem. Phys.* **61**, 5366 (1974).¹⁹D. E. Kranbuehl and P. H. Verdier, *J. Chem. Phys.* **67**, 361 (1977).²⁰D. N. Theodorou and U. W. Suter, *Macromolecules* **18**, 1206 (1985).²¹J. A. Aronovitz and D. R. Nelson, *J. Phys. France* **47**, 1445 (1986).²²M. Bishop and J. P. J. Michels, *J. Chem. Phys.* **85**, 5961 (1986).²³J. Rudnick and G. Gaspari, *Science* **237**, 384 (1987).²⁴M. Bishop and J. H. R. Clarke, *J. Chem. Phys.* **90**, 6647 (1989).

- ²⁵G. Wei and B. E. Eichinger, *J. Chem. Phys.* **93**, 1430 (1990).
- ²⁶M. Bishop, J. H. R. Clarke, A. Rey, and J. J. Freire, *J. Chem. Phys.* **94**, 4009 (1991).
- ²⁷G. Zifferer, *Macromol. Theory Simul.* **8**, 433 (1999).
- ²⁸G. Wei, *Macromolecules* **30**, 2130 (1997).
- ²⁹C. Haber, S. A. Ruiz, and D. Wirtz, *Proc. Natl. Acad. Sci. U. S. A.* **97**, 10792 (2000).
- ³⁰M. O. Steinhauser, *J. Chem. Phys.* **122**, 094901 (2005).
- ³¹Q. Liao, A. V. Dobrynin, and M. Rubinstein, *Macromolecules* **36**, 3386 (2003).
- ³²A. Jain, B. Dünweg, and J. R. Prakash, *Phys. Rev. Lett.* **109**, 088302 (2012).
- ³³M. Doi and S. F. Edwards, *The Theory of Polymer Dynamics* (Clarendon Press, Oxford, New York, 1986).
- ³⁴H. C. Öttinger, *Stochastic Processes in Polymeric Fluids* (Springer, Berlin, 1996).
- ³⁵P. Debye and E. Hückel, *Phys. Z.* **24**, 185 (1923).
- ³⁶M. J. Stevens and K. Kremer, *J. Chem. Phys.* **103**, 1669 (1995).
- ³⁷M. J. Stevens and K. Kremer, *J. Phys. II France* **6**, 1607 (1996).
- ³⁸R. B. Bird, C. F. Curtiss, R. C. Armstrong, and O. Hassager, *Dynamics of Polymeric Liquids* (John Wiley and Sons, New York, 1987), Vol. 2.
- ³⁹K. Osaki, *Macromolecules* **5**, 141 (1972).
- ⁴⁰H. C. Öttinger, *J. Chem. Phys.* **86**, 3731 (1987).
- ⁴¹J. R. Prakash, in *Advances in Flow and Rheology of Non-Newtonian Fluids*, Rheology Series edited by D. A. Siginer, D. D. Kee, and R. P. Chhabra (Elsevier Science, Amsterdam, 1999), pp. 467–517.
- ⁴²M. Kröger, A. Alba-Perez, M. Laso, and H. Öttinger, *J. Chem. Phys.* **113**, 4767 (2000).
- ⁴³P. Sunthar and J. R. Prakash, *Macromolecules* **38**, 617 (2005).
- ⁴⁴P. Sunthar and J. R. Prakash, *Europhys. Lett.* **75**, 77 (2006).
- ⁴⁵R. Prabhakar and J. R. Prakash, *J. Non-Newtonian Fluid Mech.* **116**, 163 (2004).
- ⁴⁶M. Fixman, *Macromolecules* **19**, 1204 (1986).
- ⁴⁷R. M. Jendrejack, M. D. Graham, and J. J. de Pablo, *J. Chem. Phys.* **113**, 2894 (2000).
- ⁴⁸A. Jain, P. Sunthar, B. Dünweg, and J. R. Prakash, *Phys. Rev. E* **85**, 066703 (2012).
- ⁴⁹A. Jain, “Unravelling the dynamics of semidilute polymer solutions using Brownian dynamics,” Ph.D. thesis, Monash University, 2013, <http://arrow.monash.edu.au/hdl/1959.1/901215>.
- ⁵⁰B. H. Zimm, *J. Chem. Phys.* **24**, 269 (1956).
- ⁵¹V. Degiorgio, T. Bellini, and F. Mantegazza, *Int. J. Polym. Anal. Charact.* **2**, 83 (1995).
- ⁵²K. Nishida, K. Kaji, and T. Kanaya, *J. Chem. Phys.* **115**, 8217 (2001).
- ⁵³K. Nishida, K. Kaji, and T. Kanaya, *J. Chem. Phys.* **114**, 8671 (2001).
- ⁵⁴C. G. Malmberg and A. A. Maryott, *J. Res. Natl. Bur. Stand.* **56**, RP2641 (1956).