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Description	

# C<sub>70</sub> close-packed surfaces and single molecule void-formation by local electric field through a scanning tunneling microscope tip

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A C<sub>70</sub> close-packed surface was formed by a heating of the Si surface, which is covered with C<sub>70</sub> molecules. The close-packed surface is assigned to high-temperature hexagonal close-packed phase. The stability of C<sub>70</sub> close-packed surface and formation of nanometer scale structures are studied by the application of local electric field to the close-packed surface. An application of local electric field from scanning tunneling microscope tip to the C<sub>70</sub> close-packed surface caused molecular scale evaporation. The application of local electric field near strain in the surface produced a very large void by an evaporation of more than 20 of C<sub>70</sub> molecules. © 2009 American Institute of Physics. [DOI: 10.1063/1.3075959]

Fullerenes are suitable molecules for studies with scanning tunneling microscopy (STM) because the molecular size, ~1 nm, is large enough to observe with standard STM equipments. Furthermore, fullerenes are of interests from view of nanoscale manipulation since their arrangements/orientations in bulk crystals can be easily controlled by various perturbations such as heat, light, pressure, and electric field.<sup>1</sup> The growth of C<sub>60</sub> and metal endohedral fullerenes on Si(111)-(7×7) and Si(100)-(2×1) surfaces is described as Stranski–Krastanov type.<sup>1–3</sup> On the other hand, the growth of La<sub>2</sub>@C<sub>80</sub> on hydrogen-terminated Si(100)-(2×1) surface is Volmer–Weber type.<sup>4</sup> Thus, the growth mechanism of fullerene molecules possesses diversity based on a combination of fullerene molecule and substrate.

We recently reported a formation of dark spot by removal of C<sub>60</sub> molecule from the close-packed surface by means of hole/electron injection from STM tip to the surface.<sup>5</sup> The hole/electron injection to C<sub>60</sub> molecule adjacent to the dark spot also enabled ones to move the dark spots. Furthermore, the hole/electron injection to close-packed surface of C<sub>60</sub> also induced a nanoscale chemical reaction to form C<sub>60</sub> polymer ring.<sup>6</sup> Thus, C<sub>60</sub> close-packed surface can provide a stage available for nanoscale chemical reactions as well as formations of nanoscale structures. In this letter, we report STM studies for C<sub>70</sub> molecules on Si(111)-(7×7) surface in order to clarify the nanoscale structures as well as nanoscale chemical reactions of C<sub>70</sub> molecules. Since the STM studies of C<sub>70</sub> are so far limited to those on Si(100)-(2×1) surface for Si substrate,<sup>1,7,8</sup> the growth and formation of C<sub>70</sub> close-packed layer on Si(111)-(7×7) are of interests from view of surface science and technology. Furthermore, an evaporation of C<sub>70</sub> molecules has been achieved by an application of local electric field from STM tip to C<sub>70</sub> close-packed surface.

The well-defined Si(111)-(7×7) surface was formed by the procedure described elsewhere.<sup>9</sup> The C<sub>70</sub> molecules were

deposited on the Si(111)-(7×7) surface by a thermal deposition under  $9 \times 10^{-11}$  Torr, and the temperature of Si substrate was precisely controlled in individual steps for depositions. The STM images were recorded by an observation of the constant current  $I_t$  of 0.2 nA at sample voltage  $V_s$  of 2 V under  $10^{-11}$  Torr at 295 K. The W tip was used for the STM measurement.

The STM image of C<sub>70</sub> molecules on Si(111)-(7×7) surface is shown in Fig. 1(a). 78% of C<sub>70</sub> molecules are adsorbed on A site shown in Fig. 1(b). The A site is surrounded by three Si adatoms with dangling bonds, and a dangling bond exists at the center of the site, i.e., on rest atom. Therefore, the strong chemical bonds are expected to

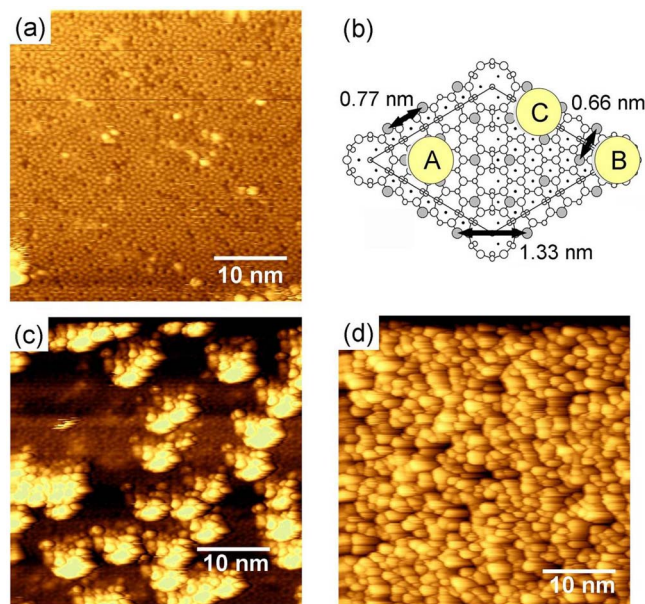


FIG. 1. (Color online) (a) STM image of C<sub>70</sub> molecules on Si(111)-(7×7) surface. Less than 0.1 ML of C<sub>70</sub> molecules are adsorbed. (b) Schematic representation of Si(111)-(7×7) surface. STM images of (c) ~0.5 ML and (d) ~2 ML of C<sub>70</sub> on Si(111)-(7×7) surface.

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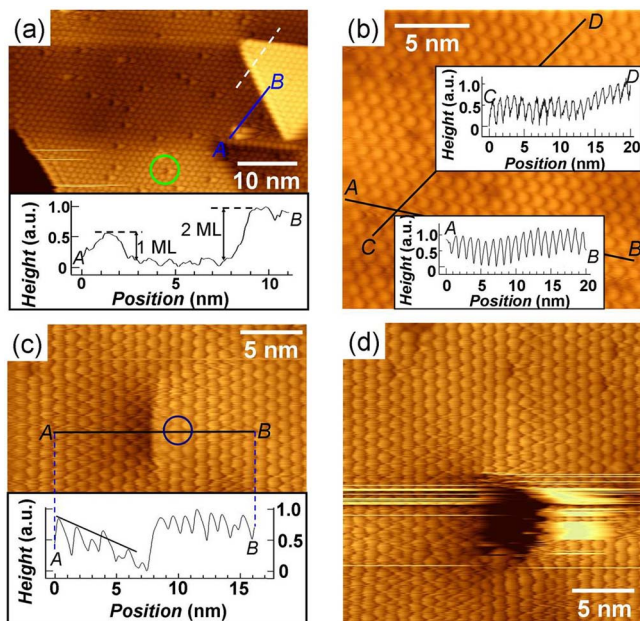


FIG. 2. (Color online) (a) STM image of close-packed surface of  $C_{70}$ . Dim spot is shown by circle and dashed line shows the alignment of  $C_{70}$  spots. Line profile of A–B is shown in the inset in (a). (b) High resolution STM image of  $C_{70}$  close-packed surface; line profiles of A–B and C–D are shown in the insets in (b). (c) STM image of  $C_{70}$  close-packed surface with strain; line profile of A–B is shown in the inset in (c). (d) STM image of  $C_{70}$  close-packed surface with a very large dark spot after electron injection; injection point is shown by circle in (c).

be formed between  $C_{70}$  and Si atoms at the A site. 4% of  $C_{70}$  molecules are adsorbed on B-site [Fig. 1(b)], which is termed “corner hole.” Furthermore, 18% of  $C_{70}$  molecules are adsorbed on C-site, which is termed “dimer line.” The fraction for  $C_{70}$  adsorption in A site is consistent with those, 72%–80%, for  $C_{60}$ ,  $Dy@C_{82}$ , and  $Ce@C_{82}$ .<sup>3,9,10</sup> However, the fractions for B and C sites (4% for B and 18% for C) are not consistent with those for  $C_{60}$  (13% for B and 7% for C),<sup>10</sup> but consistent with those for  $M@C_{82}$  (7%–9% for B and 17%–19% for C).<sup>3,9</sup> The low fraction of 4% for adsorption of  $C_{70}$  in B site may be explained by an assumption that the  $C_{70}$  molecules are adsorbed with the long axis (diameter of 1.2 nm) being parallel to Si surface and  $C_{70}$  cannot be deeply trapped into the corner hole, in the same manner as  $M@C_{82}$ .<sup>3,9</sup>

As seen from Fig. 1(c), the aggregates of  $C_{70}$  molecules are observed in the STM image of  $\sim 0.5$  monolayer (ML) of  $C_{70}$ , implying that  $C_{70}$  molecules prefer to form islands on Si surface. Such an island formation was not observed for  $C_{60}$  on Si(111)-(7 $\times$ 7) surface. Thus, the growth mechanism of  $C_{70}$  is not Stranski–Krastanov but Volmer–Weber type. Since  $C_{70}$  has a larger polarizability than  $C_{60}$ , the stronger intermolecular interaction in  $C_{70}$  than  $C_{60}$  can be expected owing to the induced dipole. This may lead to the aggregation of  $C_{70}$ , i.e., formation of islands. The STM image of  $\sim 2$  ML of  $C_{70}$  molecules are shown in Fig. 1(d). The Si surface is densely covered with  $C_{70}$  molecules.

The  $C_{70}$  molecules were subsequently deposited onto the Si surface covered with  $\sim 2$  ML of  $C_{70}$  molecules [Fig. 1(d)] heated to 373 K, and the surface was annealed at 373–393 K for 12 h after the deposition. In this process, the close-packed surface of  $C_{70}$  was observed [Fig. 2(a)]. Here it should be noted that the close-packed surface is not directly

bound to Si surface but a few  $C_{70}$  layers under the surface is formed. On the other hand, the  $C_{70}$  close-packed surface was not formed but some islands of  $C_{70}$  were observed by STM (not shown) without postannealing after the deposition. The STM image [Fig. 2(a)] showed an existence of wider  $C_{70}$  close-packed surface than  $50 \times 50$  nm<sup>2</sup> and a bright triangle  $C_{70}$  close-packed layer is formed on the wide close-packed surface. Dim spots, as shown by circle, are also observed, which can be assigned to the  $C_{70}$  polymers. The difference in height between bright triangle close-packed surface and the wide close-packed surface corresponds to 2 ML of  $C_{70}$  as seen from the line profile [inset in Fig. 2(a)]. Furthermore, the spots on the bright triangle close-packed surface are completely aligned along the spots of the wide close-packed surface, as is shown by the dashed line in Fig. 2(a). This suggests that the close packed-surface is formed by the ‘ABAB’ stacking (not ‘ABCABC’ stacking) because the arrangement of  $C_{70}$  molecule separated by 2 ML is exactly the same.

High resolution STM image of  $C_{70}$  close-packed layer is shown in Fig. 2(b). From the line profile, the  $C_{70}$ – $C_{70}$  distance was estimated to be  $1.06 \pm 0.04$  nm for A–B line and  $1.03 \pm 0.06$  nm for C–D line. The distance of 1.03–1.06 nm is consistent with those in high-temperature crystal phases,<sup>1,11,12</sup> the  $C_{70}$ – $C_{70}$  distances are 1.058 nm in high temperature hexagonal close-packed (hcp) phase (430 K) and 1.054 nm in face-centered cubic phase (383 K). The  $C_{70}$  molecules freely rotate in these phases, which are stably maintained above room temperature,<sup>11,12</sup> while the phases are also reported to be often maintained down to room temperature.<sup>13</sup> On the other hand, in low temperature phases such as distorted hcp and rhombohedral (rh), the intermolecular distance along the short axis of  $C_{70}$  molecule should be observed in STM because the molecular rotation freezes and the long axis of  $C_{70}$  molecule is oriented along [001] in distorted hcp phase and [111] in rh phase.<sup>1,13</sup> The  $C_{70}$ – $C_{70}$  distance observed by STM can be predicted for the low-temperature phases to be 1.01 nm from the lattice constants determined for these phases.<sup>11–13</sup> The  $C_{70}$ – $C_{70}$  distance observed actually by STM, 1.03–1.06 nm, is larger than that predicted for low temperature phases, showing that the STM image [Fig. 2(b)] is unambiguously assigned to high temperature phase. Further, by a consideration of the fact that this close-packed surface is consisted of “ABAB” stacking, we can assign this surface structure to high-temperature hcp phase.

The STM image of  $C_{70}$  close-packed surface often showed the structure with strain as seen from Fig. 2(c). Since the structures with strain were hardly observed in the  $C_{60}$  close-packed surface,<sup>5,6</sup> it was suggested that the  $C_{70}$  close-packed structure is unstable in comparison with  $C_{60}$  close-packed surface. This result can be reasonably understood because of lower symmetry of  $C_{70}$  ( $D_{5h}$ ) than  $C_{60}$  ( $I_h$ ); the anisotropic  $C_{70}$  shape is difficult to rotate freely to form close-packed layer and it prefers anisotropic rotation. The line profile for the area around strain is shown in the inset in Fig. 2(c). This line profile clearly shows a dislocation of  $C_{70}$  layer in the area of strain, i.e., the  $C_{70}$  layer in the left side slips under the layer in the right side.

We tried to inject electrons from STM tip fixed at the position defined by  $V_s = 2$  V and  $I_t = 1.5$  nA to the  $C_{70}$  close-packed surface. This position is closer to the surface than that for the STM-image observation ( $V_s = 2$  V and  $I_t$



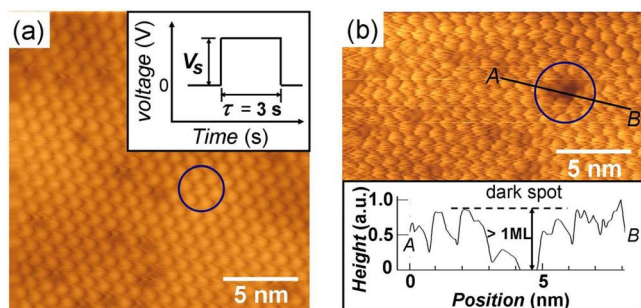


FIG. 3. (Color online) (a) STM image of close-packed surface of C<sub>70</sub>. Pulse shape used for electron injection is shown in the inset in (a). (b) STM image of C<sub>70</sub> close-packed surface with dark spot formed by electron injection; the injection point is shown by circle in (a). The line profile of A–B is shown in the inset in (b).

=0.2 nA). The shape of the applied bias-voltage pulse is schematically shown in the inset in Fig. 3(a). The STM images before/after electron injection are shown in Figs. 3(a) and 3(b), respectively. Figure 3(a) shows C<sub>70</sub> close-packed surface without dark spots, and the electrons are injected from the STM tip to the position shown by circle, at the  $V_s$  pulse of 3 V and pulse width (time)  $\tau$  of 3 s. The dark spot was observed at the part where the electrons were injected into the close-packed surface, as seen from Fig. 3(b). The dark spot can be assigned to the void formed by a removal of C<sub>70</sub> molecule on the basis of the line profile [inset in Fig. 3(b)] of the close packed surface; the line rapidly dropped by more than 1 ML in the area of dark spot. This result is consistent with that observed after electron/hole injection for the C<sub>60</sub> close-packed surface. Thus, the application of  $V_s$  pulse can cause the evaporation of C<sub>70</sub> molecule at single molecule scale.

When the electrons were injected to the position near the strain in the C<sub>70</sub> close-packed surface, the large dark spots were formed around the strain [Fig. 2(d)]; the STM image of the C<sub>70</sub> close-packed surface with strain before the electron injection is shown in Fig. 2(c). Here, it should be noticed that the fixed position of the STM tip ( $V_s=2$  V and  $I_t=1.5$  nA), applied  $V_s$  (2.8 V) and  $\tau$  (3 s) for electron injection are almost the same as those used for the single molecule evaporation [Fig. 3(b)]. Actually, more than 20 of C<sub>70</sub> molecules were removed around the strain by electron injection into the position near the strain. As a consequence, an application of local electric field to the close-packed surface may be avail-

able to obtain an indication of instability of surface in nanometer scale because the electric field application to the unstable area evaporates a number of molecules.

In conclusion, the deposition of C<sub>70</sub> molecules resulted in Volmer–Weber-type growth for Si(111)-(7×7) surface, and by an annealing of the surface at 373–393 K, the C<sub>70</sub> close-packed surface was formed over 50×50 nm<sup>2</sup>. This phase could be assigned to high-temperature hcp phase. The application of local electric field to the C<sub>70</sub> close-packed surface from STM tip resulted in evaporation of molecules. Precise control of local electric field achieved a single molecule evaporation of C<sub>70</sub>. The instability of close-packed surface of C<sub>70</sub> was supported by the existence of strains and the formation of large void around strain by application of local electric field. This study will open a way to an application of their voids toward nanometer scale mold as well as a clarification of basic behaviors of C<sub>70</sub> molecules in nanometer scale.

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