

See discussions, stats, and author profiles for this publication at: <https://www.researchgate.net/publication/230532562>

Dual Phase-Controlled Synthesis of Uniform Lanthanide-Doped NaGdF₄ Upconversion Nanocrystals Via an OA/Ionic Liquid Two-Phase System for In Vivo Dual-Modality Imaging

ARTICLE *in* ADVANCED FUNCTIONAL MATERIALS · DECEMBER 2011

Impact Factor: 11.81 · DOI: 10.1002/adfm.201101040

CITATIONS

97

READS

92

8 AUTHORS, INCLUDING:



Chunlei Zhang

Shanghai Jiao Tong University

52 PUBLICATIONS 1,131 CITATIONS

SEE PROFILE



Chenchen Bao

Shanghai Jiao Tong University

40 PUBLICATIONS 673 CITATIONS

SEE PROFILE



Guo Gao

Shanghai Jiao Tong University

50 PUBLICATIONS 971 CITATIONS

SEE PROFILE



Daxiang Cui

Shanghai Jiao Tong University

203 PUBLICATIONS 6,073 CITATIONS

SEE PROFILE

Dual Phase-Controlled Synthesis of Uniform Lanthanide-Doped NaGdF₄ Upconversion Nanocrystals Via an OA/Ionic Liquid Two-Phase System for In Vivo Dual-Modality Imaging

Meng He, Peng Huang, Chunlei Zhang, Hengyao Hu, Chenchen Bao, Guo Gao, Rong He, and Daxiang Cui*

A novel OA/ionic liquid two-phase system combining the merits of thermal decomposition method, the IL-based strategy, and the two-phase approach is introduced to synthesize high-quality lanthanide-doped NaGdF₄ upconversion nanocrystals with different crystal-phases in OA-phase and IL-phase through a one-step controllable reaction. Oil-dispersible cubic-phase NaGdF₄:Yb, Er (Ho, Tm) nanocrystals with ultra-small size (~5 nm) and monodispersity are obtained in the OA phase of the two-phase system via an IL-based reaction. More importantly, water-soluble hexagonal-phase NaGdF₄:Yb, Er nanocrystals are obtained in the same system simply by adopting an extremely facile method to complete the dual phase-transition (crystal-phase transition and OA-phase to IL-phase transition) simultaneously. The synthesized lanthanide-doped NaGdF₄ upconversion nanocrystals are effective for dual-mode UCL imaging and CT imaging in vivo.

1. Introduction

Rare earth (RE) ions doped upconversion nanoparticles (UCNPs) have attracted much attention due to their attractive merits involving narrow emission peaks, large anti-stokes shifts, and low toxicity,^[1] all of which lead to their potential applications in flat-panel displays, solid-state lasers, and especially in bio-imaging.^[2] Among these materials, NaGdF₄ nanocrystals have been considered as an excellent luminescent host matrix for various optically active Lanthanide³⁺ ions in both upconversion (UC) and downconversion (DC) luminescence.^[3] Particularly, the unique magnetic resonance (MR) properties of Gd facilitate

versatile applications as multiplexed luminescent nano-biolabels as well as MRI contrast agent. Very recently, Chen et al. reported the synthesis of NaGdF₄:Yb, Tm/NaGdF₄:Eu core-shell UCNPs and realized UC and DC luminescence simultaneously.^[4] Li et al. studied in vivo imaging of the NaGdF₄:Yb, Er/Tm with dual-modality functions of near-infrared to near-infrared (NIR-NIR) upconversion luminescence and MRI.^[5] Up to now, rare earth fluoride nanocrystals have been prepared by hydro(solvo) thermal reaction, thermal decomposition, and ionic liquids-based synthesis. Great efforts have been devoted to the synthesis of high-quality NaREF₄ nanocrystals through hydro (solvo) thermal reaction. However, it remains very difficult to synthesize hexagonal

phase NaREF₄ nanocrystals (a structure exhibiting higher UC luminescent efficiency than cubic phase structure) while maintaining small size (<50 nm). Moreover, post-synthetic modification is usually needed for the products in a hydro(solvo) thermal system for further biological applications.^[6] Up to date, thermal decomposition of trifluoroacetate precursors has been considered to be the most successful and commonly-used method to prepare hexagonal NaREF₄ UCNPs with small size (<50 nm) and high dispersity. Yet, it also exhibits some drawbacks involving rigorous experimental conditions, hazardous precursors and coordinating solvents and especially complicated surface modification of these hydrophobic products for biological compatibility.^[7] Apart from using highly toxic trifluoroacetate precursors, Chen et. al. developed a modified and relatively facile thermal decomposition method by adopting simple RE(oleate)₃ as the precursors.^[8]

Ionic liquids (ILs), due to their unique properties such as thermal and chemical stability, negligible vapor pressure, and wide electrochemical window,^[9] have emerged as a promising medium for the synthesis of inorganic nanostructures such as metal oxides, metal alloys, metal fluorides and have been regarded as a "green" alternative to the conventional organic solvents.^[10] Our group have recently successfully prepared monodisperse magnetite nanoparticles via an IL-assisted method.^[11] Besides, previously reported attempts to synthesize hydrophilic

M. He,^[†] Dr. P. Huang,^[†] C. Zhang, H. Hu, C. Bao, G. Gao, Prof. R. He, Prof. D. Cui

Department of Bio-Nano Science and Engineering
National Key Laboratory of Micro/Nano Fabrication Technology
Key Laboratory of Thin Film and Microfabrication Technology of
Department of Education
Institute of Micro-Nano Science and Technology
Shanghai Jiao Tong University
800 Dongchuan Road, Shanghai, 200240, China
E-mail: dxcui@sjtu.edu.cn

[†] M.H. and P.H. contributed equally to this work.

DOI: 10.1002/adfm.201101040

UCNPs in a one-pot IL-based reaction system, which was operated under mild conditions using innocuous reagents, provided an alternative to those conventional techniques. Kong et al.^[12] and Chen et al.^[13] have respectively obtained hexagonal and cubic $\text{NaGdF}_4\text{:Yb, Er}$ UCNPs featuring small size (<100 nm) and water-solubility in a one-step IL-based reaction. Unfortunately, the as-prepared products by IL-based method usually have poor shape-control, low dispersity and broad size distribution compared with those synthesized conventionally^[14], and the study on the synthesis of RE fluoride in an IL-based system is still at its primary stage.

In short, the well-established synthetic strategies, including hydro(solvo) thermal synthesis thermal decomposition method, as well as the newly-developed IL-based synthesis still have their respective limitations in spite of the success in some respects. In order to combine the merits of the thermal decomposition method (size and morphology control, high-quality of products) and IL-based strategy (mild experimental conditions, user-friendly reagents and direct water-solubility of products) while overcoming their respective defects, we adopt the notion of liquid-liquid two-phase approach by using RE(oleate)_3 as the precursors (the advantage of thermal decomposition method) and ILs as fluorine source (the advantage of IL-based strategy). The two-phase strategy, with its intrinsic advantages, have been proved to be successful in the synthesis of high-quality nanoparticles.^[15] Moreover, our group have recently reported a facile and controllable strategy to synthesize metallic nanocrystals in a liquid-liquid two-phase system.^[16]

Herein, we devised a novel reaction system combining the advantages of thermal decomposition method, the IL-based strategy and the liquid-liquid two-phase approach. We introduced an oleic acid (OA)/1-butyl-3-methylimidazolium tetrafluoroborate (BmimBF_4) two-phase system to fabricate high-quality $\text{NaGdF}_4\text{:Yb, Er}$ (Ho, Tm) nanocrystals with different crystal-phases in OA-phase and IL-phase through a one-step controllable reaction. We have, for the first time, obtained oil-dispersible cubic-phase $\text{NaGdF}_4\text{:Yb, Er}$ (Ho, Tm) nanocrystals with ultra-small size (~5 nm) and monodispersity via an IL-based reaction. More importantly, we have obtained water-soluble hexagonal-phase $\text{NaGdF}_4\text{:Yb, Er}$ nanocrystals in the same system simply by adopting an extremely facile method to complete the dual phase-transition (crystal-phase transition and OA-phase to IL-phase transition) simultaneously. Furthermore, in vivo dual-mode UCL imaging and CT imaging of lanthanide-doped NaGdF_4 upconversion nanocrystals were investigated.

2. Results and Discussion

In a typical procedure (see ESI[†]), RE (oleate)₃ complexes and sodium oleate were dissolved in oleic acid to form the top phase (OA-phase) of the two-phase system while BmimBF_4 formed the bottom phase (IL-phase). BF_4^- anions were partly decomposed to produce F^- , leading to the formation of $\text{NaGdF}_4\text{:Yb, Er}$ nanocrystals. In this system, a certain volume of methanol played a pivotal role in controlling the dual phase-transition of $\text{NaGdF}_4\text{:Yb, Er}$ nanocrystals. When no methanol was added, the reaction took place in OA-phase and the oleic acid capped UCNPs (OA-UCNPs) were formed. Finally, the oil-dispersible

cubic-phase $\text{NaGdF}_4\text{:Yb, Er}$ nanocrystals were obtained on the interface of the two-phase system. BmimBF_4 in IL-phase only acted as a fluorine source in this reaction. When 20 mL of methanol was added, water-soluble hexagonal-phase $\text{NaGdF}_4\text{:Yb, Er}$ nanocrystals was formed in the IL-phase and the ionic liquid capped UCNPs (IL-UCNPs) were obtained. In this case, BmimBF_4 played many important roles such as template, fluorine source and capping agent.

The sizes and morphologies of the products were characterized by transmission electron microscopy (TEM). Figure 1 shows the TEM images of oil-dispersible cubic-phase $\text{NaGdF}_4\text{: 20%Yb, 2%Er}$ nanocrystals prepared without methanol. The average diameter of these ultrasmall nanoparticles in Figure 1a was 5.05 ± 0.83 nm, which was confirmed by a detailed analysis of the particle size of around 180 nanoparticles (see Figure S2a, Supporting Information (SI)). The nanoparticles appear spherical in shape and display high uniformity and monodispersity. The diffraction rings of the selected area electron diffraction (SAED) pattern (Figure 1c) could be ascribed to the (111), (200), (220), (311), (400), and (331) planes of cubic NaGdF_4 lattice. In addition, High-resolution TEM (HR-TEM) image (Figure 1d) reveals that these ultrasmall particles were highly crystalline and shows the lattice distance of 0.28 nm, corresponding to *d* spacing for the (200) lattice plane of the cubic NaGdF_4 structure.

Figure 2 shows the TEM images of water-soluble hexagonal-phase $\text{NaGdF}_4\text{: 20%Yb, 2%Er}$ nanocrystals prepared with 20 mL of methanol. Similarly, these sphere-like particles are uniform in shape. However, these particles (Figure 2) have an average diameter of 21.04 ± 3.77 nm (see Figure S2b), larger than those oil-dispersible ones synthesized without methanol. The disparities in particle size and morphology can be attributable to the synthesis of hexagonal structure, which turned out

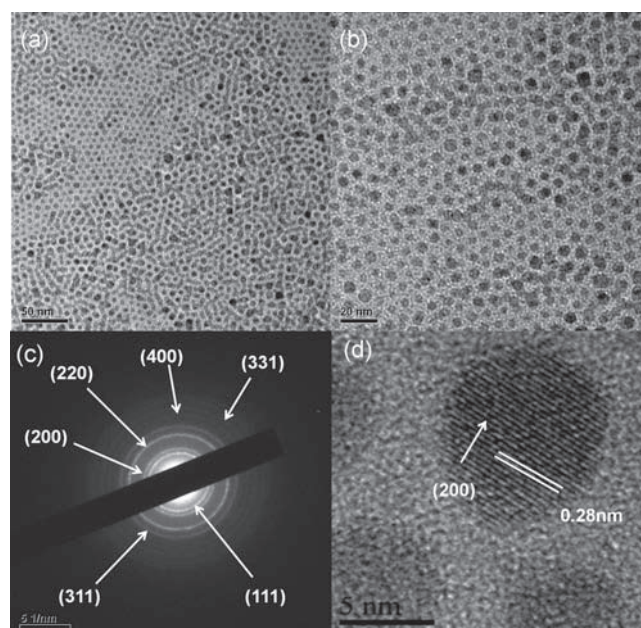


Figure 1. a,b) TEM images with different magnifications of oil-dispersible cubic-phase $\text{NaGdF}_4\text{: 20%Yb, 2%Er}$ nanocrystals prepared without methanol. c) The SAED pattern. d) The corresponding HRTEM image.

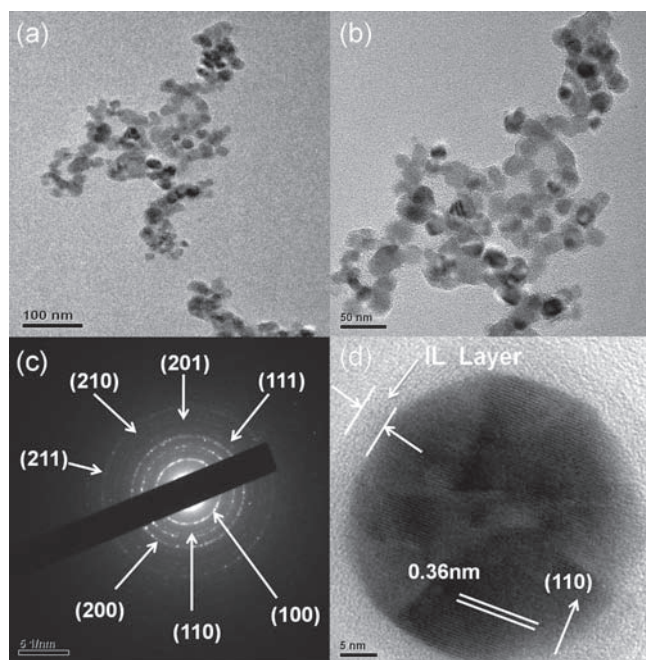


Figure 2. a,b) TEM images with different magnifications of water-soluble hexagonal phase NaGdF₄: 20%Yb, 2%Er nanocrystals prepared with 20 mL of methanol. c) The SAED pattern. d) The corresponding HR-TEM image.

to exist in larger size than their cubic-phase counterparts^[6d], as well as the change of reaction environment from OA-phase to IL-phase and to the difference in capping agents, both of which have a great impact on their nucleating and growing process. The SAED pattern in Figure 2c shows clear diffraction rings corresponding to the specific (100), (110), (200), (111), (201), (210), and (211) planes of hexagonal NaGdF₄ lattice, which indicates high crystallinity of the particles. In addition, HRTEM image (Figure 2d) shows the lattice distance of 0.36 nm, corresponding to *d* spacing for the (110) lattice plane of the hexagonal NaGdF₄ structure.

These results obviously show the crystal phase transition from cubic to hexagonal. Moreover, the change of capping agent can be demonstrated by the IL layer (marked by arrows in Figure 2d). The IL-layer absorbed on the surface of these particles made them disperse in water directly and suitable for facile surface modifications as well as further biological applications.

The crystal phase transition of the products was further characterized by X-ray diffraction (XRD) data in Figure 3. When no methanol was added (Figure 3a), the XRD pattern of NaGdF₄:20%Yb, 2%Er nanoparticles agrees well with the single-phase cubic NaGdF₄ crystals (JCPDS: 27-0697). When the volume of methanol increased to 2 mL (Figure 3b) and 10 mL (Figure 3c), their XRD patterns can be indexed as a mixture of cubic and hexagonal (JCPDS: 27-0699) phase of NaGdF₄ crystals. A gradual increase in peak intensities for hexagonal phase is observed as methanol increased. Eventually, the pure hexagonal NaGdF₄: 20%Yb, 2%Er nanocrystals were obtained when the volume of methanol reached 20 mL (Figure 3d). The peak positions and intensities accord with the single phase hexagonal NaGdF₄ crystals and no peaks of other phases or impurities are detected.

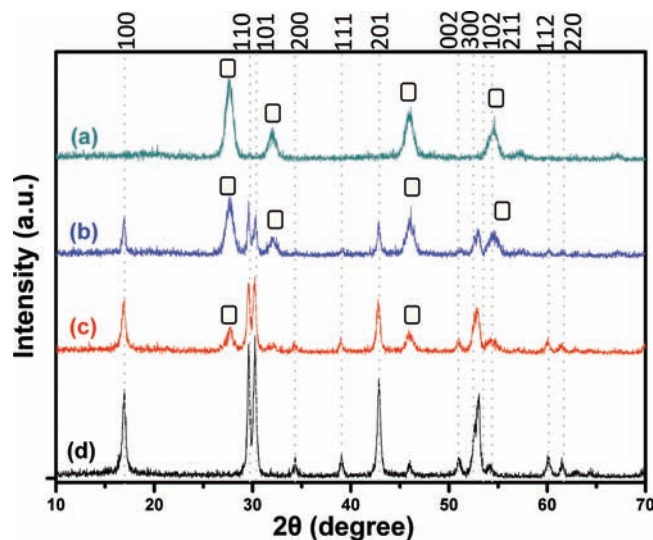


Figure 3. XRD patterns of NaGdF₄: 20%Yb, 2%Er nanocrystals when different ratios of methanol:IL (mL:mL) were added. a) 0:10; b) 2:23; c) 10:15; d) 20:5. Diffraction peaks of cubic NaGdF₄ are marked with square boxes. Different crystal planes of hexagonal NaGdF₄ lattice are marked with dotted lines.

As is shown in Figure 4, synthetic strategy 1 (A→B), marked by yellow arrow, shows the formation of cubic-phase oil-dispersible NaGdF₄:20%Yb, 2%Er nanocrystals prepared without methanol. Synthetic strategy 2 (A→C→D), marked by blue arrow, shows the formation of hexagonal-phase water-soluble NaGdF₄:20%Yb, 2%Er nanocrystals when methanol was added. Photographs of real products clearly show their final distribution (see Fig. S3, ESI). In the OA/IL two-phase system, RE (oleate)₃ complexes and sodium oleate were dissolved in oleic acid as precursors. BmimBF₄ in IL-phase partly decomposed under high temperature and produced F⁻, leading to the formation of NaREF₄ nanocrystals.^[17] When no methanol was added, the OA-phase and IL-phase were immiscible to each other even under high pressure and temperature. The reaction only took place on the interface area of the two-phase system. Due to the strong chelating capacity of oleic acid with RE³⁺ ions, the surface of the nanocrystals was covered by a layer of oleic acid, which prevented the particles from agglomerating and rendered them hydrophobic. Thus, we prepared oil-dispersible products in OA-phase. In this case, BmimBF₄ only acted as a fluorine source. When a certain volume of methanol was added, the methanol was soluble in both oleic acid and BmimBF₄ and therefore provided a “passageway” for RE³⁺ ions to transfer from OA-phase to IL-phase. Then RE³⁺ ions were surrounded by BF₄⁻ anions and Bmim⁺ cations layer-by-layer due to coulombic or coordination interactions. When heated continuously, BF₄⁻ anions were degraded, resulting in the formation of NaREF₄ nanocrystals. Since this reaction was carried out in IL-phase, BmimBF₄ not only functioned as a fluorine source, but also as a template and capping/ coordinating agent. The nucleating and growing process of the particles were shaped by the IL layer coordinated and absorbed on the surface, which ultimately decided their sizes and morphologies. Moreover, owing to the hydrophilic properties of

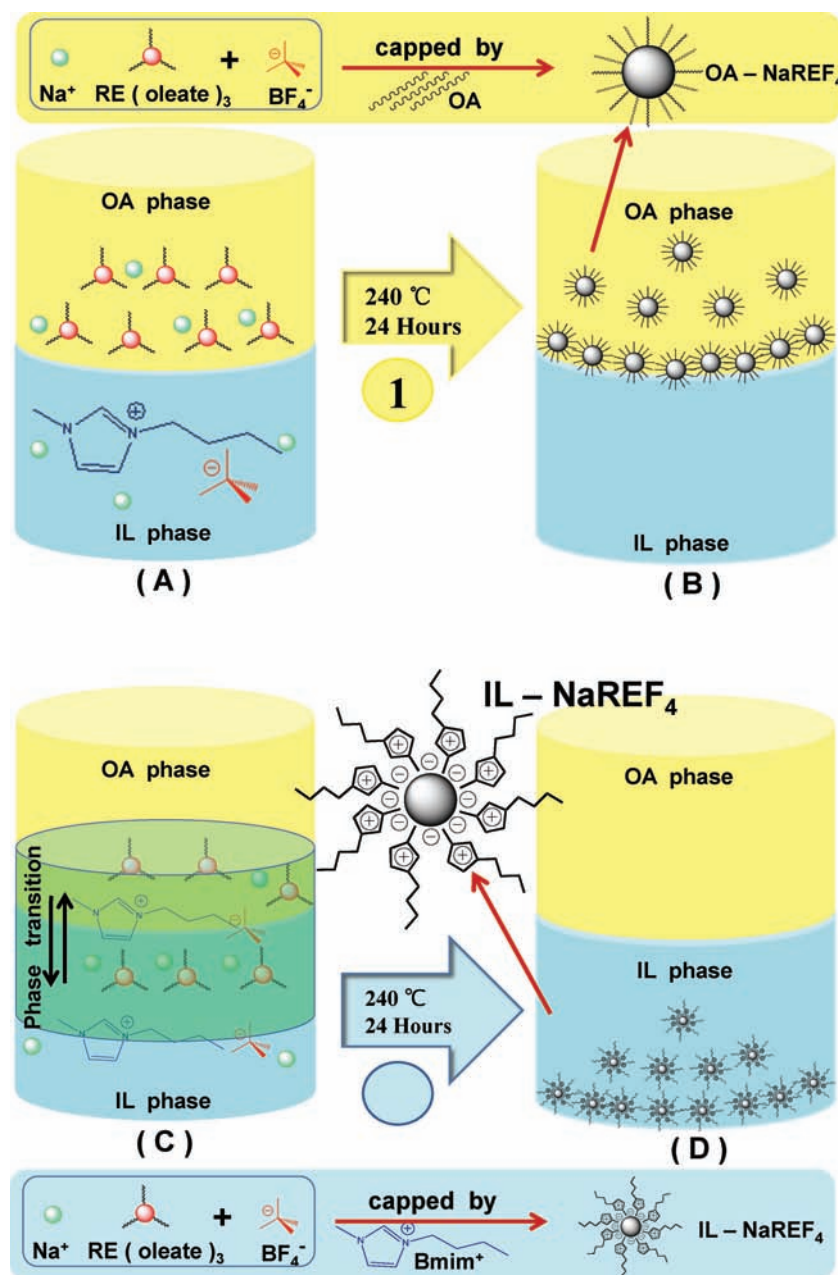


Figure 4. Schematic diagram showing the mechanism for the formation NaGdF₄ nanocrystals.

BmimBF₄ layer, these particles can be directly dispersed in water, which make them feasible for biological applications. The surface capping agents of products in OA-phase and IL-phase were further characterized by FTIR (see Figure S1).

In the case of crystal-phase transition from cubic to hexagonal, it can be reasonably speculated that the crystal-phase transition can be ascribed to the elevated concentration of F⁻ ions and the change in reaction environment of RE³⁺ ions. In a pure OA/IL two-phase system, the formation of NaGdF₄:Yb, Er nanocrystals only occurred in the interfacial area of the two-phase system due to the low inter-solubility of OA and BmimBF₄. It is rational to believe that the majority of RE³⁺ precursors were coordinated

by oleic acid due to the strong coordinating capacity of OA. The interactions between BF₄⁻ ions and RE³⁺ precursors were quite limited and the effective concentration of F⁻ ions released from the cleavage of B-F bond in the formation of NaGdF₄ was quite low. So the synthesis of cubic phase products was favored. However, in a methanol-induced OA/IL two-phase system, the introduction of methanol successfully transferred all RE³⁺ precursors into IL phase and RE³⁺ ions were surrounded by Bmim⁺ and BF₄⁻ ions through coordination and coulombic interactions. Therefore, the interactions between BF₄⁻ ions and RE³⁺ precursors were enhanced and effective concentration of F⁻ ions was elevated, which could effectively reduce the energy barrier from cubic-phase to hexagonal-phase.^[18] On the other hand, the selective absorption of Bmim⁺ and BF₄⁻ ions on the surface of the particles would facilitate their anisotropic growth in the IL phase and thus favored the fabrication of hexagonal phase structure.

It should be noted that the one-step synthesis of water-soluble products facilitate their further modification (including Stöber-based silica coating and layer-by-layer assembly of PSS and PAH) under mild conditions to enhance stability and bio-compatibility. The corresponding TEM images (Figure S4,S5, S1) and FT-IR spectra (Figure S6) are shown in supporting information.

The upconversion luminescence (UCL) spectra of Er³⁺, Ho³⁺, Tm³⁺ doped nanocrystals was shown in Figure 5. The hexagonal (β) phase water-soluble products prepared with 20 mL methanol presented much higher luminescence intensities than those cubic (α) phase oil-dispersible ones synthesized without methanol. The main possible reason could be that the IL-UCNPs have a relatively larger size and smaller ratio surface, and thus less surface defects which are usually fluorescence quenchers.^[19] On the other hand, a crystal phase change from cubic to hexagonal NaREF₄ offers at least an order of magnitude promotion of UC efficiency

relative to its cubic counterparts.^[20] As is shown in Figure 5a There are three emission peaks at around 520, 538 and 652 nm for NaGdF₄:20%Yb, 2%Er, corresponding to the ²H_{11/2}-⁴I_{15/2}, ⁴S_{3/2}-⁴I_{15/2} and ⁴F_{9/2}-⁴I_{15/2} transitions of Er³⁺ respectively. In Figure 5b, the NaGdF₄:20%Yb, 2%Ho samples exhibit four distinct Ho³⁺ emission bands. The blue emissions at around 484 nm is originated from ⁵F₃-⁵I₈ transition; the intense green emission at around 539 nm accounts for the ⁵F₄, ⁵S₂-⁵I₈ transition; the weaker red and near-infrared (NIR) luminescence was ascribed to the ⁵F₅-⁵I₈, and ⁵F₄, ⁵S₂-⁵I₇ transitions of Ho³⁺ respectively. In the case of Tm³⁺ doped products (Figure 5c, d), five weak emission bands at around 445, 474, 538, 652 and

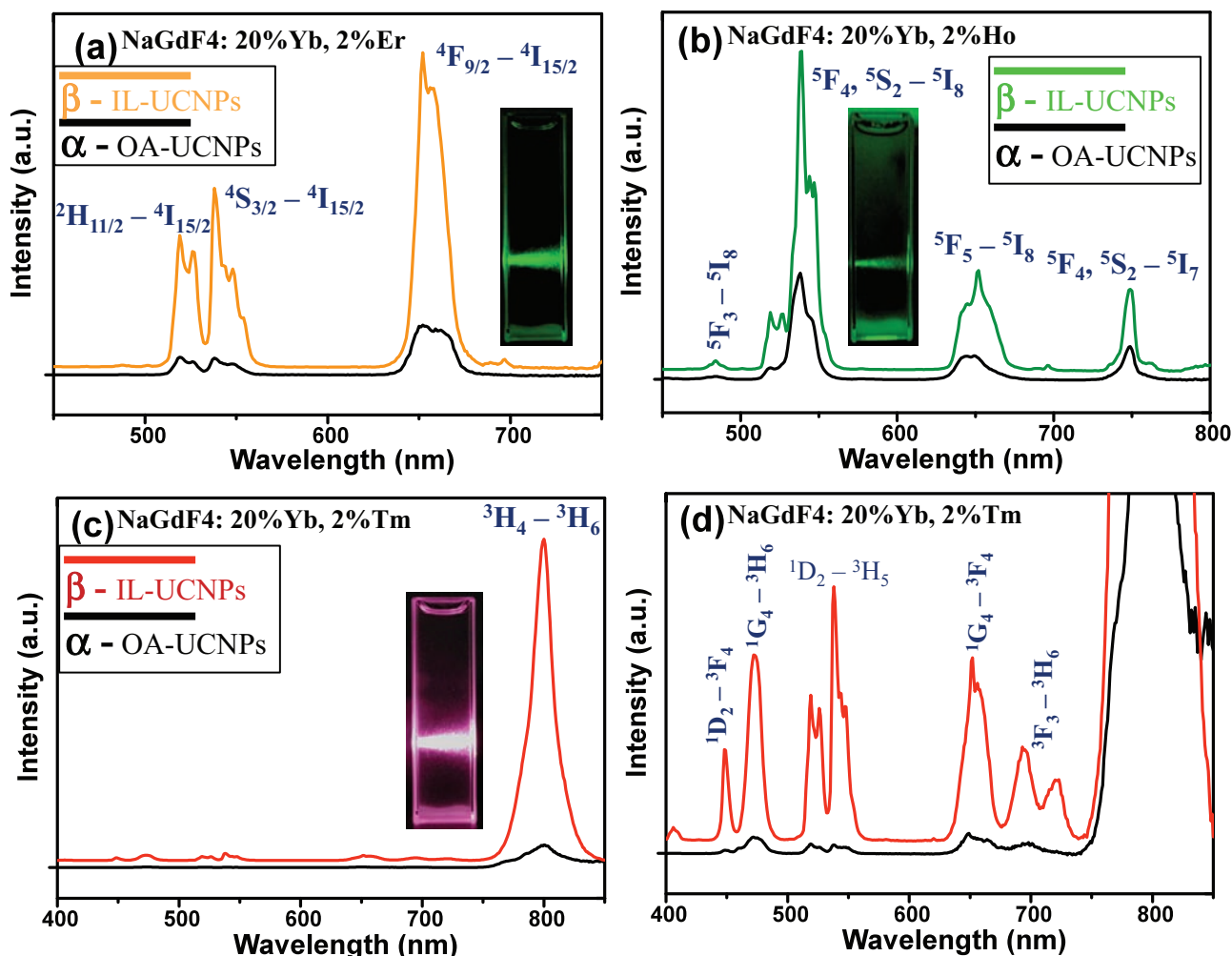


Figure 5. Upconversion luminescence spectra of a) NaGdF₄:20%Yb, 2%Er. b) NaGdF₄:20%Yb, 2%Ho. c) NaGdF₄:20%Yb, 2%Tm. d) the amplified UC spectrum of (c). Inset: The UC luminescent photographs of water-soluble β - NaGdF₄ UCNPs dispersed in water under 980 nm excitation.

693 nm was caused by 1D_2 - 3F_4 , 1G_4 - 3H_6 , 1D_2 - 3H_5 , 1G_4 - 3F_4 , and 3F_3 - 3H_6 transitions of Tm³⁺ respectively. It is striking that the predominant NIR emission band peaked at around 800 nm, which is assigned to the 3H_4 - 3H_6 transition, is much stronger than all other emission bands, indicating that radiative deactivation of the upconverted energy predominantly occurs through NIR luminescence, which is favorable for bioimaging.^[21]

To demonstrate the feasibility of lanthanide-doped NaGdF₄ upconversion nanocrystals for in vivo dual-modality imaging, the water-soluble hexagonal-phase NaGdF₄:20%Yb, 2%Er was selected as a contrast reagent. **Figure 6** presents the in vivo whole-body images of nude mice without and with NaGdF₄:Yb, Er (100 μ L 1 mg mL⁻¹ per animal). The mice were imaged using a modified IVIS Lumina XR in which excitation was provided by the CW infrared laser at 980 nm. As is shown in Figure 6a, a clearly distinguished UCL signal was observed at the corresponding subcutaneous injection site, whereas no significant

luminescence signal was observed in the control mice. Meanwhile, the subcutaneous injection site displayed an enhancer positive-contrast than other soft tissues (Figure 6b), which is attributed to that NaGdF₄:20%Yb, 2%Er induces a strong X-ray attenuation. The X-ray attenuation coefficient is determined by the atomic number and electron density, the higher the atomic number and electron density, the higher the attenuation

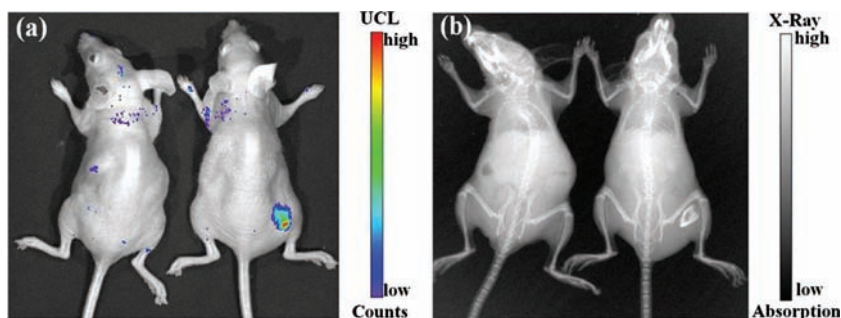


Figure 6. In vivo upconversion luminescence imaging (a) and X-ray imaging (b) of mice after subcutaneous injection (left) without and (right) with NaGdF₄: Yb, Er.

coefficient. The atomic number and electron density of Gd (64 and 7.895 g cm^{-3} , respectively) are much higher than those of the currently used iodine (53 and 4.9 g cm^{-3}).^[22] The results indicated that the synthesized lanthanide-doped NaGdF₄ upconversion nanocrystals are effective for dual-mode UCL imaging and X-ray imaging in vivo.

Based on the absorption of X-ray, computed tomography (CT) can provide complementary anatomical information. Currently, CT contrast agents are typically low molecular weight and rapidly cleared by body in clinic.^[22] Nanoparticle agents exhibited superior prolonged presence in the blood pool, which may make them suitable agents for vascular CT.^[23] Therefore, we investigate the feasibility of lanthanide-doped NaGdF₄ upconversion nanocrystals for in vitro and in vivo CT imaging. Figure 7a displays the CT images in the range of $0.5\text{--}10 \text{ mg mL}^{-1}$ of lanthanide-doped NaGdF₄ upconversion nanocrystals. As the mass concentration of lanthanide-doped NaGdF₄ upconversion nanocrystals increased, the CT signal intensity continuously increased resulting in brighter images. To further investigate the CT contrast effect, the attenuation values (HU) of lanthanide-doped NaGdF₄ upconversion nanocrystals at different concentrations were measured by a clinical CT (SIEMENS Spirit CT 31159, German). As shown in Figure 7b, HU as a function of the lanthanide-doped NaGdF₄

upconversion nanocrystals concentration display a well-correlated linear relationship ($R^2 = 0.9999$), described by the following typical equation: $Y = 14.37 - 5.77X$.

For in vivo CT imaging, the lanthanide-doped NaGdF₄ upconversion nanocrystals were injected into mice. The mice were imaged using the same CT as in vitro test. Figure 8 presents the in vivo whole-body image of nude mice with lanthanide-doped NaGdF₄ upconversion nanocrystals ($100 \mu\text{L}$ of 10 mg mL^{-1} per site). The corresponding injection site of mice displayed a clearly distinguished CT signal. Meanwhile, HU values of injection site is more than the values of other tissue (such as bone, muscle), indicating that the synthesized nanocrystals is effective for CT imaging in vivo. To the best of knowledge, it should be the first to be reported. Compared with traditional CT images, the as-prepared lanthanide-doped NaGdF₄ upconversion nanocrystals exhibit distinct advantages, however, so far it is not clarified well to compare with analogous ones using other Ln³⁺-based NPs, which is under our further investigation. Keeping in mind that Gd³⁺ is a paramagnetic relaxation agent used in MRI, the multi-model in vivo imaging combining optical, CT and MR imaging is underway.

3. Conclusions

In summary, we have successfully synthesized high quality RE³⁺ doped NaGdF₄ nanocrystals with different properties in a newly-developed facile OA/IL two-phase system. Ultrasmall oil-dispersible cubic-phase OA-capped products with high uniformity and monodispersity were prepared in OA-phase. While water-soluble hexagonal-phase IL-capped products with uniform shape and size were synthesized in IL-phase. A certain volume of methanol has played a role of paramount importance in controlling the simultaneous dual phase-transition. In addition, we have demonstrated that the hydrophilic coatings and strong UCL of these IL-UCNPs were suited for UCL imaging and CT imaging in vivo. The multi-functional imaging system combining different modality such as optical, CT and

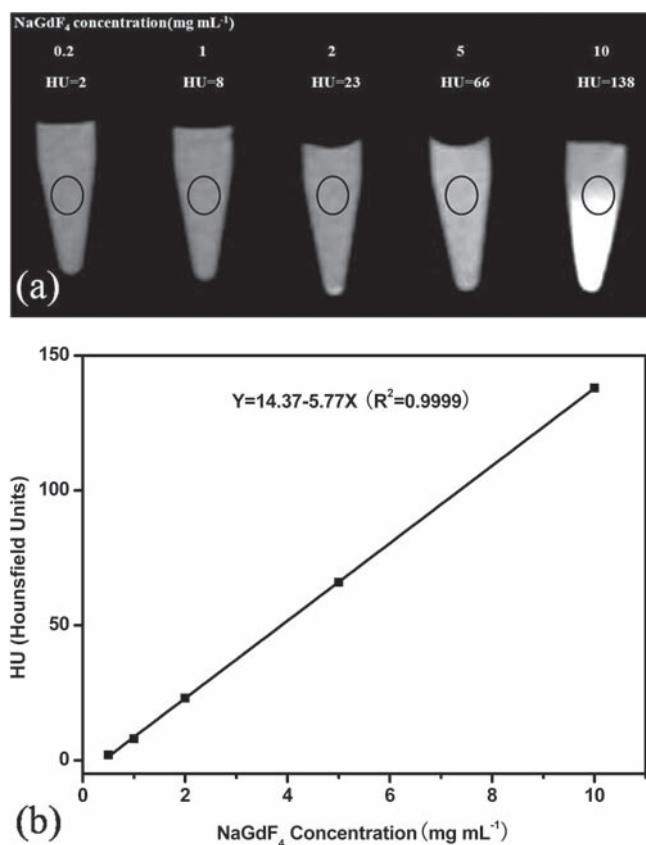


Figure 7. a) In vitro CT images of lanthanide-doped NaGdF₄ upconversion nanocrystals suspended in PBS. The concentration (mg mL^{-1}) in each sample is provided at the top of the respective images. b) CT attenuation (HU) of lanthanide-doped NaGdF₄ upconversion nanocrystals at various concentrations.

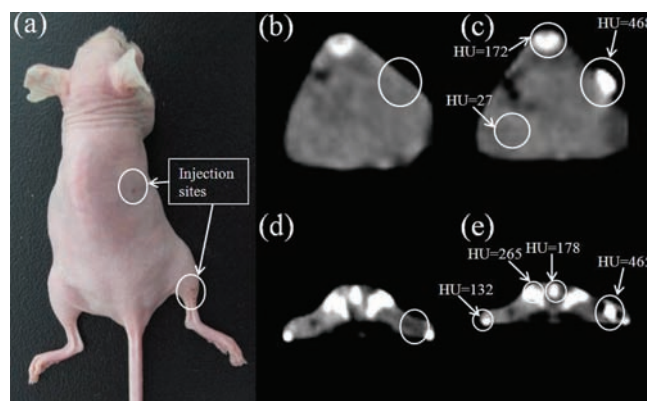


Figure 8. In vivo CT images of mice after subcutaneous injection with lanthanide-doped NaGdF₄ upconversion nanocrystals suspended in PBS. a) The photograph of the injected mouse; b,c) the transverse image of the back, the HU value of the injection site is 468; d,e) the transverse image of the buttock, the HU value of the injection site is 465.

MR imaging could bring novel opportunities to the next generation of probes for the detection and imaging of biological targets in vivo.

4. Experimental Section

Materials: All the chemicals were of analytical grade and were used without further purification. Deionized water was used throughout. Rare earth oxides Gd_2O_3 (99.99%), Yb_2O_3 (99.99%), Er_2O_3 (99.99%), Ho_2O_3 (99.99%), and Tm_2O_3 (99.99%) were purchased from Aladdin Chemistry Co. Ltd. Sodium oleate, oleic acid (OA), methanol were obtained from Sinopharm Chemical Reagent Co. 1-butyl-3-methylimidazolium tetrafluoroborate (BmimBF₄) was purchased from Henan Lihua Pharmaceutical Co. Ltd.

Synthesis of Rare Earth Oleate Complexes: RE (oleate)₃ complexes were prepared by previously reported methods.^[24,25] One mmol of RE (oleate)₃[Gd: Yb: Er(Ho, Tm)] = [78: 20: 2] and a stoichiometric amount of sodium oleate was dissolved in 7 mL of oleic acid at elevated temperature under nitrogen atmosphere with vigorous magnetic stirring to form a homogeneous solution.

Synthesis of Oil-Dispersible Cubic-Phase NaGdF₄: Yb, Er (Ho, Tm) Nanocrystals in OA-phase: The as-prepared OA-phase containing Na⁺ and RE (oleate)₃ was transferred into a 50-mL Teflon-lined autoclave which contained 10 mL of BmimBF₄. The two phase system was treated at 240 °C for 24 hours. After cooling to room temperature naturally, the precipitates were found on the interface of the two phase system. Then they were washed by cyclohexane and ethanol several times under ultrasonic conditions and were collected through centrifugation at 9500 rpm. Finally, the products are dried at 70 °C under vacuum for 12 hours.

Synthesis of Water-Soluble Hexagonal-Phase NaGdF₄: Yb, Er (Ho, Tm) Nanocrystals in IL-Phase: The synthetic procedure was the same as that used to prepare the oil-dispersible products except that 20 mL of methanol was added to the two-phase system and that the volume of BmimBF₄ was reduced to 5 mL. In contrast, the all the precipitates were found in the ionic-liquid phase.

Characterization: The sizes and morphologies of the products were characterized by transmission electron microscopy (TEM) using a JEM 2010 microscopy (JEOL, Japan) and High-resolution TEM (HRTEM) on JEM 2100F microscopy (JEOL, Japan) with an acceleration voltage of 200 kV. Powder X-ray diffraction (XRD) analysis was performed on a Dmax-r C X-ray diffractometer (Rigaku, Japan) with Cu K α radiation at 1.540 Å° at a scanning rate of 6°/min in the 2 θ range from 10°–70°. Fourier transform infrared (FT-IR) spectra were measured by EQUINOX 55 spectrometer (Bruker, Germany) from samples on KBr pellets. Upconversion fluorescence spectra were recorded on Fluorolog-3 spectrofluorometer (Jobin Yvon, France) at room temperature.

Animal Experiments and In Vivo Dual-Modality Imaging: Male nude mice weighing 20–22 g were purchased from Shanghai SLAC Laboratory Animal Co. Ltd (Shanghai, China). All animal procedures were in agreement with institutional animal use and care regulations. In vivo dual-modality UCL imaging and X-ray imaging were performed with a modified IVIS Lumina XR (Xenogen Corporation-Caliper, Alameda, CA, USA) using an external 0–5 W adjustable CW infrared laser (980 nm, Shanghai Connet Fiber Optics Co., China) as the excited source. Nude mice were randomized into two groups of three animals per group including control group and subcutaneous injection group. After mice were anesthetized (with 100 μL 10% chloral hydrate), for subcutaneous injection group, the water-soluble hexagonal-phase NaGdF₄: Yb, Er aqueous solution (100 μL 1 mg mL⁻¹) in 0.9% NaCl saline solution was subcutaneously injected into the right rear flank area. Then the optical/X-ray whole-body images of mice were recorded on IVIS Lumina XR instrument. For in vivo CT imaging, the NaGdF₄: Yb, Er aqueous solution (100 μL 10 mg mL⁻¹ per site) in PBS was subcutaneously injected into the back of mice, and was intramuscularly injected into the right hind leg of mice, respectively. All CT scans were performed using a clinical CT (SIEMENS Spirit CT 31159, German).

Supporting information

Supporting Information is available from the Wiley Online Library or from the author.

Acknowledgements

This work is supported by National 863 Hi-tech Project of China (No. 2007AA022004), National Key Basic Research Program (973 Project) (No. 2010CB933901 and 2011CB933100), Important National Science & Technology Specific Projects (2009ZX10004-311), National Natural Scientific Fund (No. 20803040), New Century Excellent Talent of Ministry of Education of China (NCET-08-0350), Shanghai Science and Technology Fund (No. 10XD1406100), and Shanghai Jiao Tong University Innovation Fund for Postgraduates.

Received: May 10, 2011

Published online: September 22, 2011

- [1] a) F. Wang, X. G. Liu, *Chem. Soc. Rev.* **2009**, 38, 976; b) F. Wang, D. Banerjee, Y. S. Liu, X. Y. Chen, X. G. Liu, *Analyst* **2010**, 135, 1839; c) H. S. Mader, P. Kele, S. M. Saleh, O. S. Wolfbeis, *Curr. Opin. Chem. Biol.* **2010**, 14, 582; d) L. D. Carlos, R. A. S. Ferreira, V. De Zea Bermudez, S. J. L. Ribeiro, *Adv. Mater.* **2009**, 21, 509; e) J. C. G. Bünzli, *Chem. Rev.* **2010**, 110, 2729.
- [2] a) C. Wang, L. Cheng, Z. Liu, *Biomaterials* **2011**, 32, 1110; b) L. Cheng, K. Yang, S. Zhang, M. W. Shao, S. T. Lee, Z. Liu, *Nano Res.* **2010**, 3, 722; c) C. X. Liu, *Nano Biomed Eng.* **2009**, 1, 1; d) D. X. Cui, Y. D. Han, Z. M. Li, H. Song, K. Wang, R. He, B. Liu, H. L. Liu, C. C. Bao, P. Huang, J. Ruan, F. Gao, H. Yang, H. S. Cho, Q. S. Ren, D. L. Shi, *Nano Biomed Eng.* **2009**, 1, 61; e) P. Huang, Z. Li, J. Lin, D. P. Yang, G. Gao, C. Xu, L. Bao, C. L. Zhang, K. Wang, H. Song, H. Y. Hu, D. X. Cui, *Biomaterials* **2011**, 32, 3447; f) Z. Li, P. Huang, X. Zhang, J. Lin, S. Yang, B. Liu, F. Gao, P. Xi, Q. Ren, D. X. Cui, *Mol. Pharm.* **2009**, 7, 94; g) P. Huang, C. Xu, J. Lin, C. Wang, X. Wang, C. Zhang, X. Zhou, S. Guo, D. X. Cui, *Theranostics*, **2011**, 1, 240.
- [3] a) F. Wang, X. Fan, M. Wang, Y. Zhang, *Nanotechnology* **2007**, 18, 025701; b) C. H. Liu, H. Wang, X. R. Zhang, D. P. Chen, *J. Mater. Chem.* **2009**, 19, 489; c) J. C. Boyer, J. Gagnon, L. A. Cuccia, J. A. Capobianco, *Chem. Mater.* **2007**, 19, 3358; d) P. Ptacek, H. Schäfer, K. Kömpe, M. Haase, *Adv. Funct. Mater.* **2007**, 17, 3843; e) Y. I. Park, J. H. Kim, K. T. Lee, K. S. Jeon, H. B. Na, J. H. Yu, H. M. Kim, N. Lee, S. H. Choi, S. I. Baik, H. Kim, S. P. Park, B. J. Park, Y. W. Kim, S. H. Lee, S. Y. Yoon, I. C. Song, W. K. Moon, Y. D. Suh, T. Hyeon, *Adv. Mater.* **2009**, 21, 4467.
- [4] Y. S. Liu, D. T. Tu, H. M. Zhu, R. F. Li, W. Q. Luo, X. Y. Chen, *Adv. Mater.* **2010**, 22, 3266.
- [5] J. Zhou, Y. Sun, X. X. Du, L. Q. Xiong, H. Hu, F. Y. Li, *Biomaterials* **2010**, 31, 3287.
- [6] a) Z. G. Chen, H. L. Chen, H. Hu, M. X. Yu, F. Y. Li, Q. Zhang, Z. G. Zhou, T. Yi, C. H. Huang, *J. Am. Chem. Soc.* **2008**, 130, 3023; b) Z. Q. Li, Y. Zhang, *Angew. Chem., Int. Ed.* **2006**, 45, 7732; c) X. Wang, Y. D. Li, *Nature* **2005**, 437, 121; d) F. Wang, Y. Han, C. S. Lim, Y. H. Lu, J. Wang, J. Xu, H. Y. Chen, C. Zhang, M. H. Hong, X. G. Liu, *Nature* **2010**, 463, 1061.
- [7] a) H. X. Mai, Y. W. Zhang, R. Si, Z. G. Yan, L. S. Sun, L. P. You, C. H. Yan, *J. Am. Chem. Soc.* **2006**, 128, 6426; b) J. C. Boyer, F. Vetrone, L. A. Cuccia, J. A. Capobianco, *J. Am. Chem. Soc.* **2006**, 128, 7444; c) G. S. Yi, G. M. Chow, *Chem. Mater.* **2007**, 19, 341.
- [8] C. H. Liu, H. Wang, X. Li, D. P. Chen, *J. Mater. Chem.* **2009**, 19, 3546.
- [9] a) P. Hapiot, C. Lagrost, *Chem. Rev.* **2008**, 108, 2238; b) K. Ding, Z. Miao, Z. Liu, Z. Zhang, B. Han, G. An, S. Miao, Y. Xie, *J. Am. Chem. Soc.* **2007**, 129, 6362; c) H. J. Ryu, L. Sanchez, H. A. Keul, A. Raj, M. R. Bockstaller, *Angew. Chem., Int. Ed.* **2008**, 47, 7639.

- [10] a) M. Antonietti, D. Kuang, B. Smarsly, Y. Zhou, *Angew. Chem. Int. Ed.* **2004**, *43*, 4988; b) C. Burda, X. B. Chen, R. Narayanan, M. A. El-Sayed, *Chem. Rev.* **2005**, *105*, 1025; c) K. Biswas, C. N. R. Rao, *Chem.-Eur. J.* **2007**, *13*, 6123; d) D. S. Jacob, L. Bitton, J. Grinblat, I. Felner, Y. Koltypin, A. Gedanken, *Chem. Mater.* **2006**, *18*, 3162; e) Y. J. Zhu, W. W. Wang, R. J. Qi, X. L. Hu, *Angew. Chem. Int. Ed.* **2004**, *43*, 1410; f) Y. Jiang, Y. J. Zhu, G. F. Cheng, *Crystal Growth & Design* **2006**, *6*, 2174; g) W. W. Wang, Y. J. Zhu, *Crystal Growth & Design* **2005**, *5*, 505; h) C. Lorbeer, J. Cybinska, A. V. Mudring, *Chem. Commun.* **2010**, 63, 571; i) N. V. Prondzinski, J. Cybinska, A. V. Mudring, *Chem. Commun.* **2010**, 46, 4393.
- [11] H. Y. Hu, H. Yang, P. Huang, D. X. Cui, Y. Q. Peng, J. C. Zhang, F. Y. Lu, J. Lian, D. L. Shi, *Chem. Commun.* **2010**, 46, 3866.
- [12] X. M. Liu, J. W. Zhao, Y. J. Sun, K. Song, Y. Yu, C. Du, X. G. Kong, H. Zhang, *Chem. Commun.* **2009**, 43, 6628.
- [13] C. Zhang, J. Chen, *Chem. Commun.* **2010**, 46, 592.
- [14] C. X. Li, J. Lin, *J. Mater. Chem.* **2010**, *20*, 6831.
- [15] a) D. Pan, Q. Wang, L. An, *J. Mater. Chem.* **2009**, *19*, 1063; b) J. Yang, J. Ying, *Nat. Mater.* **2009**, *8*, 683.
- [16] P. Huang, J. Lin, Z. M. Li, H. Y. Hu, K. Wang, G. Gao, R. He, D. X. Cui, *Chem. Commun.* **2010**, 46, 4800.
- [17] a) C. N. R. Rao, S. R. C. Viekchand, K. Biswas, A. Govindaraj, *Dalton Trans.* **2007**, 3728; b) C. Zhang, J. Chen, Y. Zhou, D. Li, *J. Phys. Chem. C*, **2008**, *112*, 10083.
- [18] P. Ghosh, A. Patra, *J. Phys. Chem. C*, **2008**, *112*, 3223.
- [19] F. Wang, J. Wang, X. G. Liu, *Angew. Chem. Int. Ed.* **2010**, *49*, 7456.
- [20] S. Heer, K. Kompe, H. U. Güdel, M. Hasse, *Adv. Mater.* **2004**, *16*, 2102.
- [21] G. Y. Chen, Y. G. Zhang, G. Somesfalean, Z. G. Zhang, *Appl. Phys. Lett.* **2006**, *89*, 163105; b) G. Y. Chen, T. Y. Ohulchanskyy, R. Kumar, H. Agren, P. N. Prasad, *ACS nano*, **2010**, *6*, 3163.
- [22] R. Popovtzer, A. Agrawal, N. A. Kotov, A. Popovtzer, J. Balter, T. E. Carey, R. Kopelman, *Nano Lett.* **2008**, *8*, 4593.
- [23] S. M. Janib, A. S. Moses, J. A. MacKay, *Adv Drug Deliv Rev.*, **2010**, *62*, 1052.
- [24] Y. Wei, F. Q. Lu, X. R. Zhang, D. P. Chen, *Chem. Mater.* **2006**, *18*, 5733.
- [25] J. Park, K. J. An, Y. S. Hwang, J. G. Park, H. J. Noh, J. Y. Kim, J. H. Park, N. M. Hwang, T. Hyeon, *Nat. Mater.* **2004**, *3*, 891.