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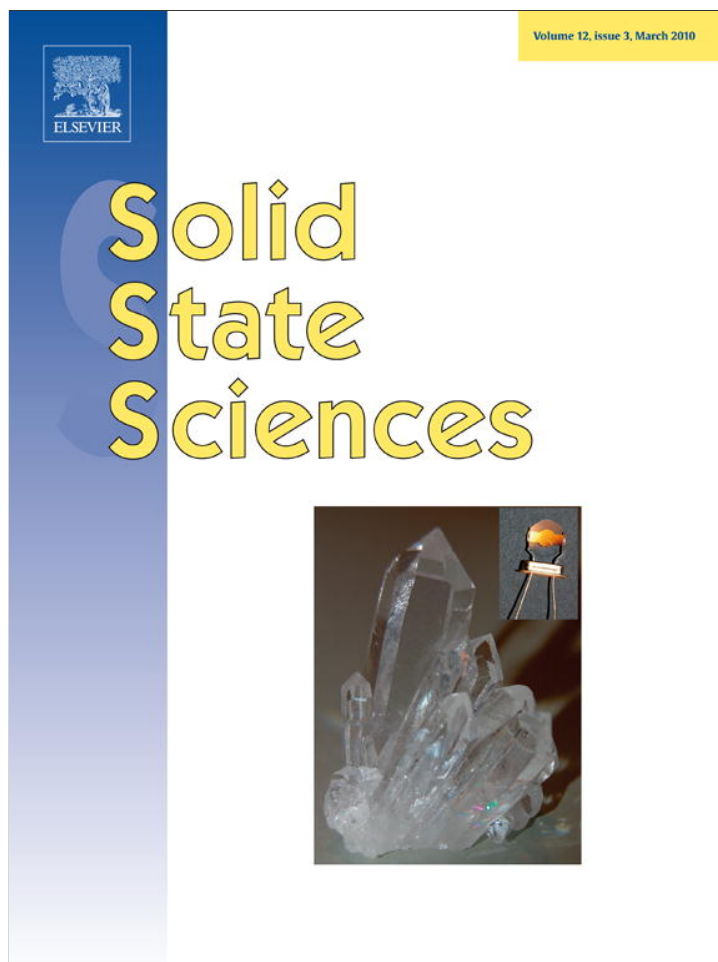


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# Structural, electronic and magnetic properties of cementite-type $\text{Fe}_3\text{X}$ ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) by first-principles calculations<sup>☆</sup>

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## ABSTRACT

Using first-principles technique, the crystal structure of cementite-type  $\text{Fe}_3\text{N}$  is predicted. The average magnetic moment ( $M_s$ ) of cementite-type  $\text{Fe}_3\text{N}$  is also predicted as  $1.4929 \mu_B/\text{atom}$ . The  $M_s$  of  $\text{Fe}_3\text{N}$  is bigger than that of  $\text{Fe}_3\text{C}$ , but smaller than that of  $\text{Fe}_3\text{B}$ . Fe  $M_s$  between two different Fe sites in  $\text{Fe}_3\text{N}$  are different ( $2.0541$  and  $2.0139 \mu_B$ ), which indicates that Fe  $M_s$  are sensitive to the local short-range order in the cementite-type crystal. The  $M_s$  of B, C and N are  $-0.3525$ ,  $-0.2474$  and  $-0.1102 \mu_B/\text{atom}$  in  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ), respectively. The chemical bonds of  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) take on metallicity, covalence, and ionicity. The ionicity of  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) strengthens and the covalence of Fe–X weakens, going from  $\text{Fe}_3\text{B}$ ,  $\text{Fe}_3\text{C}$  to  $\text{Fe}_3\text{N}$ .

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## 1. Introduction

As the cementite-type  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) with orthorhombic structure are the most important strengthening phases in carbon and alloy steels, the carburizing, boriding and nitriding technology are applied widely to improve the surface properties of the steels [1–3]. In some cases,  $\text{Fe}_3\text{B}$  and  $\text{Fe}_3\text{N}$  are also the important composition phases in the other ferromagnetic materials [4–6]. Although the structural, electronic, and magnetic properties of  $\text{Fe}_3\text{N}$  with hexagonal structure have been investigated [6,7], there are few reports on the properties of cementite-type  $\text{Fe}_3\text{N}$ . The investigations on the electronic structure and magnetic properties of  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) are a significant contribution to clarify the relationship between them and predict the properties of the cementite-type  $\text{Fe}_3\text{N}$ .

The present study aims to examine the electronic and magnetic properties of cementite-type  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) and to clarify the relationship between them by using the first-principles method. As

a result, the crystal structural, electronic and magnetic properties of the cementite-type  $\text{Fe}_3\text{N}$  are predicted.

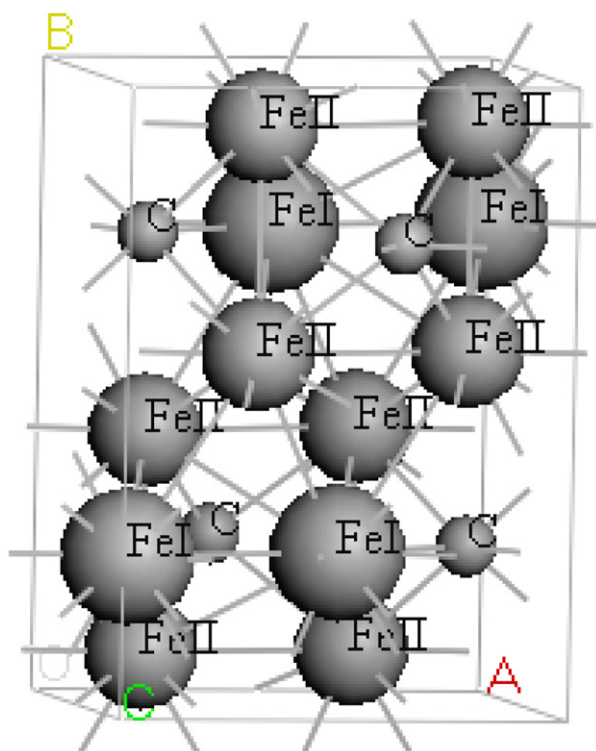
## 2. Crystal structure and calculation details

All calculations were performed for cementite-type  $\text{Fe}_3\text{X}$  (where  $\text{X} = \text{B}, \text{C}, \text{N}$ ) using CASTEP code [8] based on the Density-Function Theory (DFT) [9,10]. The electron–ion interactions are described by using the Vanderbilt ultrasoft pseudopotential (USPP) [11]. The exchange and correlation terms are treated by Generalized Gradient Approximation (GGA) in the scheme of Revised Perdew–Burke–Eruzer (RPBE) [12]. It is needed to mention that the used USPP should be compatible with the special type of GGA. For the Brillouin-Zone sampling, the Monkhorst–Pack Scheme [13] was adopted.

In the calculations of  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ), we used  $5 \times 4 \times 6$  mesh of special k-points and the kinetic energy cutoff ( $E_{\text{cut}}$ )  $E_{\text{cut}} = 500 \text{ eV}$  for the GGA calculations. Spin polarization is considered at each calculation. Each calculation was considered convergence when the maximum force on the atom was below  $0.006 \text{ eV \AA}^{-1}$  and the maximum displacement between cycles was below  $2 \times 10^{-4} \text{ \AA}$ . The cementite-type  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) crystallize in the orthorhombic space group Pnma (SP No. 62) with four formula units ( $Z = 4$ ) per unit cell, where four iron atoms (FeI) in “special” positions, eight

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**Fig. 1.** Crystal structure of cementite-type  $\text{Fe}_3\text{X}$ , showing the orthorhombic space lattice and the sixteen atoms basis.

iron atoms ( $\text{Fe}_{\text{II}}$ ) are in “general” positions, and four nonmetal atoms (B, C, N) in the interstices [14]. A three-dimensional drawing of the unit cell is shown in Fig. 1.

### 3. Results and discussions

#### 3.1. Structural properties of cementite-type $\text{Fe}_3\text{X}$ ( $\text{X} = \text{B}, \text{C}, \text{N}$ )

The ground state properties of the cementite-type  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) are investigated from their total energy, which is calculated as a function of volume. When it is according to the Murnaghan equation of state [15], the equilibrium lattice constants and atomic positions can be obtained (see Table 1). It can be found that the GGA values of the lattice constants match fairly well with the experimental ones for  $\text{Fe}_3\text{B}$  [16] and  $\text{Fe}_3\text{C}$  [17]. The deviations between the experimental and theoretical values of cell volume are 0.32% and 0.28% for  $\text{Fe}_3\text{B}$  and  $\text{Fe}_3\text{C}$ , respectively. Based on the same method

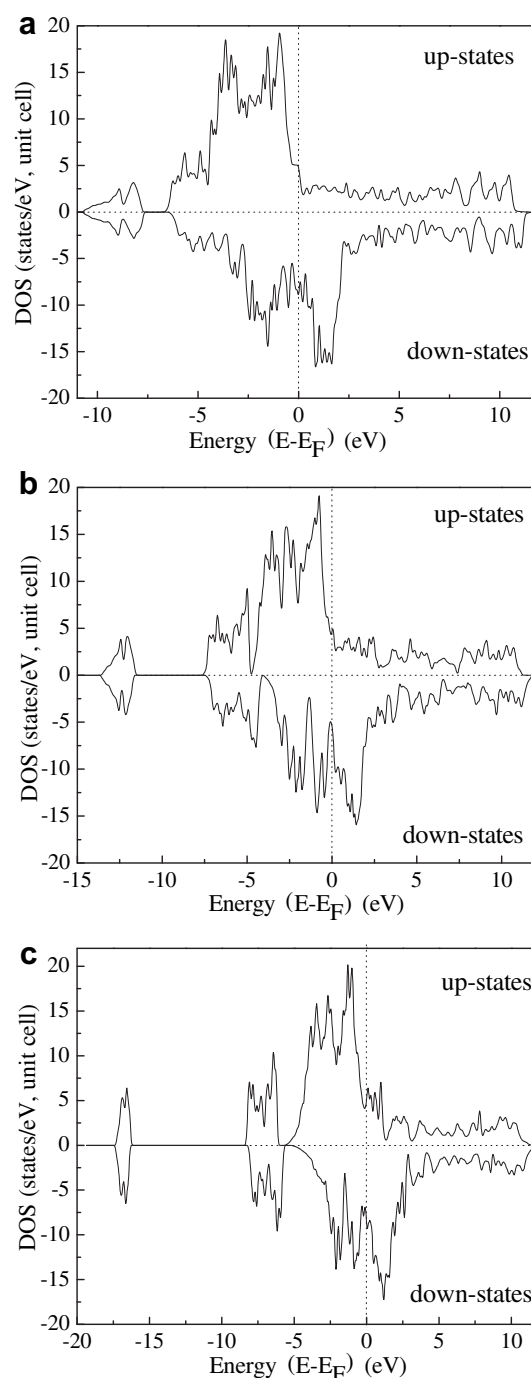
**Table 1**  
Experimental and calculated lattice constants  $a_0$ ,  $b_0$ ,  $c_0$  (Å),  $v_0$  (Å<sup>3</sup>),  $\rho$  (g/cm<sup>3</sup>) and  $E_{\text{cell}}$  (eV) of  $\text{Fe}_3\text{B}$ ,  $\text{Fe}_3\text{C}$  and  $\text{Fe}_3\text{N}$ .

	$\text{Fe}_3\text{B}$ (Ref. [16])	$\text{Fe}_3\text{C}$ (Ref. [17])	$\text{Fe}_3\text{N}$
$a$	5.437 (5.433)	5.053 (5.082)	4.955
$b$	6.710 (6.656)	6.775 (6.733)	7.124
$c$	4.387 (4.454)	4.504 (4.521)	4.506
$\text{Fe}_\text{I}$	0.02041, 0.25000, 0.88216	0.03887, 0.25000, 0.83490	0.05881, 0.25000, 0.81467
$\text{Fe}_\text{II}$	0.17557, 0.44364, 0.35267	0.17654, 0.43136, 0.33139	0.17374, 0.42347, 0.31483
$\text{X}$	0.38182, 0.25000, 0.07487	0.37688, 0.25000, 0.06203	0.38201, 0.25000, 0.06536
$V$	160.055 (161.066)	154.258 (154.696)	159.067
$\rho$	7.401 (7.355)	7.731 (7.709)	7.581
$E_{\text{total}}$ (cell)	−10 706.46	−11 013.76	−11 477.78

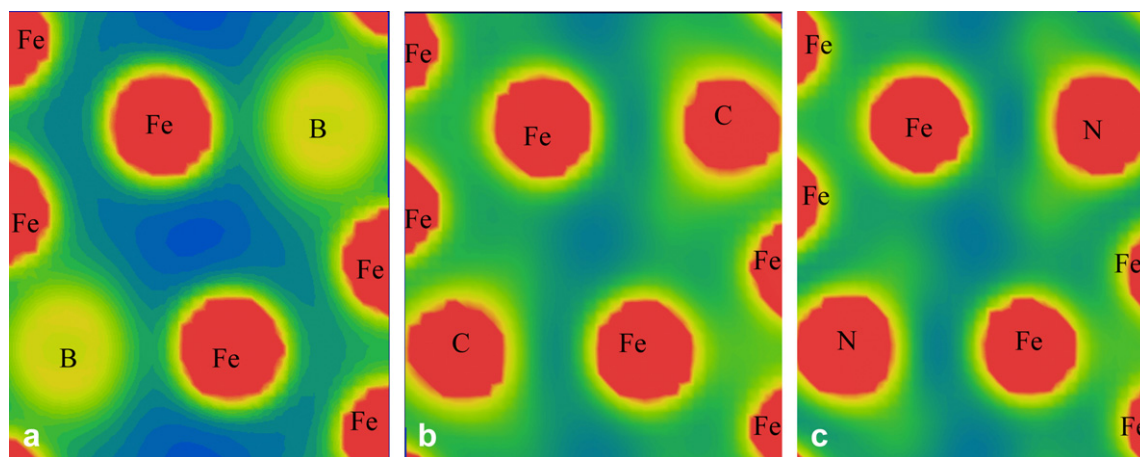
the crystal structure of the cementite-type  $\text{Fe}_3\text{N}$  is predicted (shown in Table 1).

#### 3.2. Electronic properties of cementite-type $\text{Fe}_3\text{X}$ ( $\text{X} = \text{B}, \text{C}, \text{N}$ )

Considering the results obtained from the electronic structure of cementite-type  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) at equilibrium, the electronic properties of them from the plots of DOSs (density of states) are discussed. The site- and spin-projected DOSs are shown in Fig. 2 for  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) at theoretical equilibrium lattice constants. It can be seen that for  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ), there are three regions: the lowest valence band, the upper valence band, and the conduction



**Fig. 2.** Calculated spin-polarized total density of states of (a)  $\text{Fe}_3\text{B}$ , (b)  $\text{Fe}_3\text{C}$ , (c)  $\text{Fe}_3\text{N}$ .



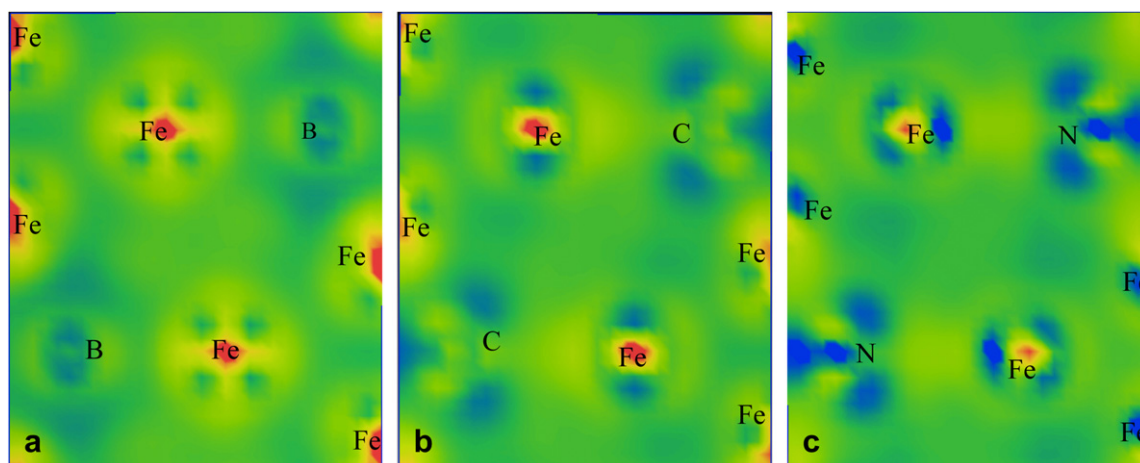
**Fig. 3.** Total electron density distribution map of the plane with Fe and X atoms of  $\text{Fe}_3\text{X}$  plotted from 0.01 (blue) to 1 (red)  $\text{e} \cdot \text{\AA}^{-3}$ , electron density values larger than 1 are not shown (a)  $\text{Fe}_3\text{B}$ , (b)  $\text{Fe}_3\text{C}$ , (c)  $\text{Fe}_3\text{N}$ . (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

unoccupied states. The lowest valence band ranging consists of a mixture of X (B, C, N) 2s and a small contribution from s, p and d states of metal atoms Fe. The upper valence band mainly constituted with the hybridization of X (B, C, N) 2p and metal atoms (Fe) 3d. Comparing the up with down densities, it can be seen that the up and down states are not symmetric for them, and  $\text{Fe}_3\text{X}$  (X = B, C, N) have magnetic characters. The magnetic characters of them are manifest in the two last parts from Fig. 2. Actually, the lowest valence band is almost symmetric, and it is near to the Fermi level that the up and down states are noticeably dissimilar for them. The energy gaps between the lowest valence band and the upper valence band are 1, 4 and 7.5 eV for cementite-type  $\text{Fe}_3\text{B}$ ,  $\text{Fe}_3\text{C}$  and  $\text{Fe}_3\text{N}$ , respectively. This shows that the chemical bonds of  $\text{Fe}_3\text{X}$  (X = B, C, N) take on ionicity and the ionicity of Fe–X strengthens from Fe–B, Fe–C to Fe–N. The main reason of the ionicity of Fe–X is the difference in electro-negativity between the comprising elements (Fe and X). In addition, no energy gap near to the Fermi level can be seen, and this indicates a metallic nature of  $\text{Fe}_3\text{X}$  (X = B, C, N). The bonding characters of the constituting elements can be described as a mixture of covalent–ionic and, due to the d-resonance in the vicinity of the Fermi level, partly metallic.

In Figs. 3 and 4, the electron density distribution map of the plane with Fe and X (B, C, N) atoms is plotted in two ways, the total

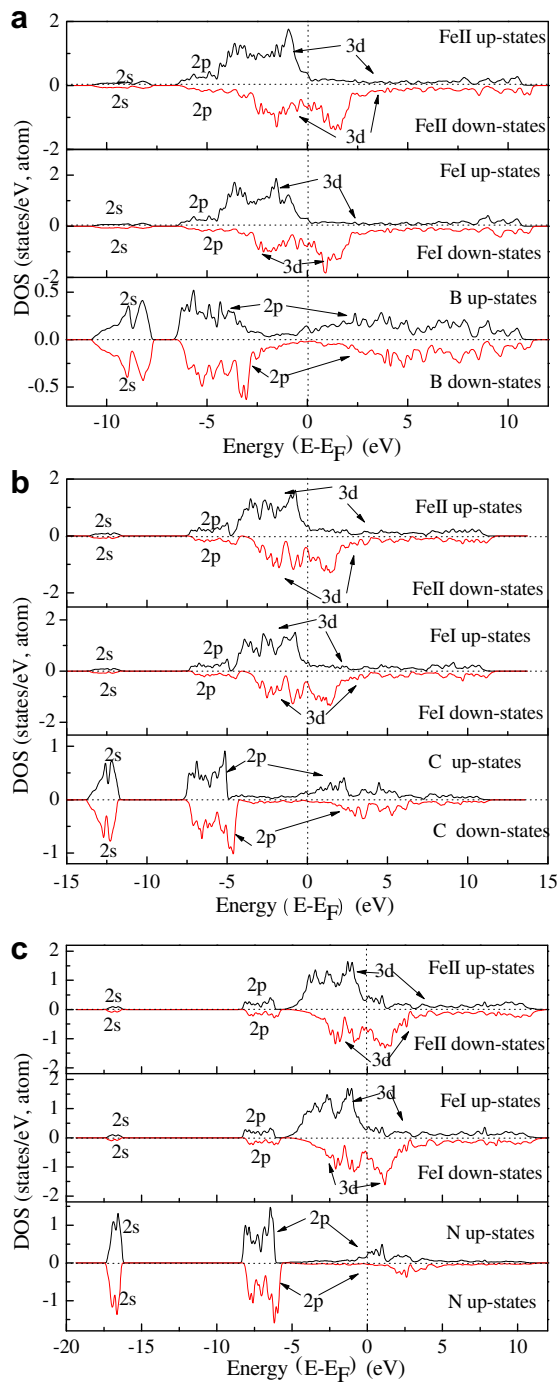
density map and the electron density difference map. The core regions of X (X = B, C, N) and Fe have the largest density from Fig. 3, which are mainly due to ionic core orbits. The covalent bonds of Fe–X can be found in Fig. 3, which were immersed in a metallic Fermi free electron gas. The covalence of Fe–X grows, going from Fe–N, Fe–C to Fe–B. The delocalized electron clouds represent the metallic bonds between the metal atoms and the highest energy level near the Fermi surface being occupied [18]. The electron density difference was determined as  $\Delta\rho = \{\rho_{\text{crystal}} - \sum \rho_{\text{at}}\}$ , where  $\rho_{\text{crystal}}$  and  $\rho_{\text{at}}$  are the valence electron densities for  $\text{Fe}_3\text{X}$  (X = B, C, N) and the corresponding free atoms, respectively. It can be seen that the increment of valence electrons is concentrated on the X (X = B, C, N) atoms and the elongated contours correspond to the p-like orbits of X (X = B, C, N). In the interstitial regions, the increment of delocalized electrons is attributed to the metallic bonds. From Figs. 3 and 4, it is concluded that the chemical bonds of  $\text{Fe}_3\text{X}$  (X = B, C, N) take on metallicity and covalence, and the covalence of Fe–B is strongest and that of Fe–C is weakest in them.

The differences in electro-negativity between Fe and X result in the charge transfer in  $\text{Fe}_3\text{X}$ . For explaining the charge transfer and the ionicity of Fe–X, the Mulliken populations for  $\text{Fe}_3\text{X}$  were calculated. From the Mulliken population analysis for cementite ( $\text{Fe}_3\text{C}$ ), the charges of C,  $\text{Fe}_\text{I}$  and  $\text{Fe}_\text{II}$  atoms distribute to  $-0.69$ ,



**Fig. 4.** Electron density difference map of the plane with Fe and X atoms of  $\text{Fe}_3\text{X}$  plotted from  $-0.2$  (blue) to  $0.1$  (red)  $\text{e} \cdot \text{\AA}^{-3}$  (a)  $\text{Fe}_3\text{B}$ , (b)  $\text{Fe}_3\text{C}$ , (c)  $\text{Fe}_3\text{N}$ . (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)





**Fig. 5.** Calculated site- and spin-projected partial density of states of (a)  $\text{Fe}_3\text{B}$ , (b)  $\text{Fe}_3\text{C}$ , (c)  $\text{Fe}_3\text{N}$ .

+0.19 and +0.25, respectively [19]. As for  $\text{Fe}_3\text{B}$ , the charges of B, FeI and FeII atoms are  $-0.73$ ,  $+0.25$  and  $+0.24$ , respectively; as for  $\text{Fe}_3\text{N}$ , the charges of N, FeI and FeII atoms distribute to  $-0.67$ ,  $+0.19$  and  $+0.24$ . The values of charge transfer for  $\text{Fe}_3\text{C}$ ,  $\text{Fe}_3\text{B}$  and  $\text{Fe}_3\text{N}$  are 0.69, 0.73 and 0.67, respectively. The calculated lengths of the Fe–B bonds are 2.08327, 2.10503 and 2.13917 Å respectively, and the corresponding overlapping-population values are 0.35, 0.31 and 0.25 in  $\text{Fe}_3\text{B}$ . The calculated lengths of the Fe–C bonds are 1.99094, 2.00195 and 2.00849 Å respectively, and the corresponding overlapping-population values are 0.30, 0.32 and 0.34 in  $\text{Fe}_3\text{C}$ . The calculated lengths of the Fe–N bonds are 1.95992,

**Table 2**

Calculated effective charges  $Q^*$ , spin magnetic moments  $M_s$ , and DOS at the Fermi level (states/eV spin unit cell) for  $\text{Fe}_3\text{B}$ ,  $\text{Fe}_3\text{C}$  and  $\text{Fe}_3\text{N}$ .

	$\text{Fe}_3\text{B}$		$\text{Fe}_3\text{C}$		$\text{Fe}_3\text{N}$	
	Fe (4c)	Fe (8d)	Fe (4c)	Fe (8d)	Fe (4c)	Fe (8d)
<b>Fe <math>Q_i^*</math> (electron)</b>						
s	0.2367	0.2329	0.2332	0.2316	0.2269	0.2194
p	0.3037	0.3198	0.3504	0.3152	0.3484	0.3180
d	4.5617	4.4453	4.4080	4.3736	4.4115	4.3787
Sum	5.1021	4.9980	4.9616	4.9204	4.9868	4.9161
<b>Fe <math>Q_i^*</math> (electron)</b>						
s	0.2375	0.2429	0.2498	0.2470	0.2473	0.2380
p	0.3685	0.4019	0.4474	0.3914	0.4570	0.4100
d	2.0391	2.2097	2.2459	2.2872	2.2284	2.2542
Sum	2.6451	2.8545	2.9431	2.9256	2.9327	2.9022
$Q^* = Q_i^* + Q_i^*$	7.7473	7.8525	7.9047	7.8460	7.9195	7.8183
$M_s = Q_i^* - Q_i^*$	2.4570	2.1435	2.0185	1.9948	2.0541	2.0139
<b>B <math>Q_i^*</math> (electron)</b>						
s	0.5576	0.7274				
p	1.1283	1.4944				
Sum	1.6859	2.2218				
<b>C <math>Q_i^*</math> (electron)</b>						
s	0.6205	0.7518				
p	1.4179	1.7174				
Sum	2.0384	2.4692				
<b>N <math>Q_i^*</math> (electron)</b>						
s					0.8439	
p					2.0422	
Sum					2.8861	
$Q^* = Q_i^* + Q_i^*$	3.7243	4.6910			5.6620	
$M_s = Q_i^* - Q_i^*$	−0.3525	−0.2474			−0.1102	
$N(E_F)$ (cell)	4.7885	4.5866			4.9285	
$N(E_F)^{\dagger}$ (cell)	8.8610	5.3696			8.2713	
$N(E_F)$ (total)	13.6495	9.9562			13.1998	

1.96349 and 1.97664 Å respectively, and the corresponding overlapping-population values are 0.30, 0.25 and 0.28 in  $\text{Fe}_3\text{N}$ . This indicates that the strong covalent bonding states exist in  $\text{Fe}_3\text{X}$  ( $X = \text{B}, \text{C}, \text{N}$ ). The lengths of Fe–Fe bonds in  $\text{Fe}_3\text{B}$ ,  $\text{Fe}_3\text{C}$  and  $\text{Fe}_3\text{N}$  are 2.45748–2.95550 Å, 2.45834–2.67174 Å and 2.47077–2.82024 Å, respectively. The variations of population values are 0.03–1, 0.06–1 and 0.03–1 in  $\text{Fe}_3\text{B}$ ,  $\text{Fe}_3\text{C}$  and  $\text{Fe}_3\text{N}$ , respectively. There are no negative overlap population values of Fe–Fe bonds in  $\text{Fe}_3\text{X}$  ( $X = \text{B}, \text{C}, \text{N}$ ), which indicates no repulsion force among these atoms.

### 3.3. Magnetic properties of cementite-type $\text{Fe}_3\text{X}$ ( $X = \text{B}, \text{C}, \text{N}$ )

Fig. 5 shows the site- and spin-projected Partial Densities of States (PDOSs) for  $\text{Fe}_3\text{X}$  ( $X = \text{B}, \text{C}, \text{N}$ ), respectively. For each case, the up and down total densities at each atom are shown separately. From Fig. 5, it can be found that the spin DOSs of Fe atom in  $\text{Fe}_3\text{B}$ ,  $\text{Fe}_3\text{C}$ ,  $\text{Fe}_3\text{N}$  are different. The effective charges  $Q^*$ , their spin components and the spin magnetic moments were calculated according to the Mulliken scheme for each atom in  $\text{Fe}_3\text{X}$  ( $X = \text{B}, \text{C}, \text{N}$ ) from Fig. 5, which are listed in Table 2.

The average magnetic moments of the  $\text{Fe}_3\text{B}$ ,  $\text{Fe}_3\text{C}$  and  $\text{Fe}_3\text{N}$  are 1.5979, 1.4402 and 1.4929  $\mu_B$ /atom, respectively. The results of  $\text{Fe}_3\text{C}$  and  $\text{Fe}_3\text{B}$  are similar to that in the literatures [19,20]. The average magnetic moment of cementite– $\text{Fe}_3\text{N}$  (1.4929  $\mu_B$ /atom) is similar to that of  $\epsilon$ - $\text{Fe}_3\text{N}$  (1.44  $\mu_B$ /atom) [7]. The differences of  $M_s$  between  $\text{Fe}_3\text{X}$  mainly result from that of the spin density difference of Fe and X ( $X = \text{B}, \text{C}, \text{N}$ ) in the three crystals (see Fig. 5). For cementite-type  $\text{Fe}_3\text{B}$ , the values of Fe  $M_s$  are 2.4570 and 2.1435  $\mu_B$  for FeI and FeII, respectively. For cementite– $\text{Fe}_3\text{C}$ , the values of Fe  $M_s$  are 2.0185 and 1.9948  $\mu_B$  for FeI and FeII, respectively. For cementite-type  $\text{Fe}_3\text{N}$ , the values of Fe  $M_s$  are 2.0541 and 2.0139  $\mu_B$  for FeI and FeII, respectively. The magnetic character of Fe atom comes from the spin of 3d electrons. The spin density of Fe of  $\text{Fe}_3\text{B}$  is bigger in  $E_F$  (the Fermi level) than that of  $\text{Fe}_3\text{C}$  and  $\text{Fe}_3\text{N}$ . The values of atom  $M_s$  can be

obtained from the deviation between the up spin occupied state and down spin occupied state. The spin magnetic moment of Fe for  $\text{Fe}_3\text{B}$  is bigger than that for  $\text{Fe}_3\text{N}$  and  $\text{Fe}_3\text{C}$ , so the Ms of  $\text{Fe}_3\text{B}$  is stronger than  $\text{Fe}_3\text{N}$  and  $\text{Fe}_3\text{C}$ . The Ms of  $\text{Fe}_3\text{C}$  is smaller than that of  $\text{Fe}_3\text{N}$ , because the Ms of N ( $-0.1102 \mu_{\text{B}}$ ) is bigger than that of C atom ( $-0.2474 \mu_{\text{B}}$ ) in the crystal. The Fe Ms between two different Fe sites in cementite-type  $\text{Fe}_3\text{X}$  are also different, which indicates that the Fe Ms are sensitive to the local short-range order in the crystals. Similar difference was also found in the literature [19], while the values of moments in Fe are different. The corresponding  $Q^*$  for Fe atoms in  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) are similar, but that for B, C and N are different (see Table 2). The corresponding  $Q^*$  for N is bigger than that for B and C in  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ). This indicates that the ionicity of cementite-type  $\text{Fe}_3\text{N}$  is stronger than  $\text{Fe}_3\text{C}$  and  $\text{Fe}_3\text{B}$ , and that of  $\text{Fe}_3\text{B}$  is weakest in them. This result accords with the DOSs analyses.

#### 4. Conclusion

In summary, a completely theoretical analysis of the structural, electronic and magnetic properties of cementite-type  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) has been presented using first-principles technique. The internal positions of atoms within the unit cell were optimized and the ground state properties such as lattice parameter and the final enthalpy of cementite-type  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) were calculated. The calculated equilibrium structural parameters of  $\text{Fe}_3\text{C}$  and  $\text{Fe}_3\text{B}$  are in agreement with the experimental results. Using the same method the crystal structure of the cementite-type  $\text{Fe}_3\text{N}$  is predicted. From the calculated results, it is confirmed that the bonds of cementite-type  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) are the unusual mixtures of metallicity,

covalence, and ionicity. The ionicity of cementite-type  $\text{Fe}_3\text{N}$  is the strongest and the covalence of cementite-type  $\text{Fe}_3\text{B}$  is the strongest in them. The cementite-type  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ) are ferromagnetic phases, and the average magnetic moment of cementite-type  $\text{Fe}_3\text{N}$  is predicted as  $1.4929 \mu_{\text{B}}/\text{atom}$ . The Ms of  $\text{Fe}_3\text{N}$  is bigger than that of  $\text{Fe}_3\text{C}$ , but smaller than that of  $\text{Fe}_3\text{B}$ . Fe moments between two different Fe sites are different in cementite-type  $\text{Fe}_3\text{X}$  ( $\text{X} = \text{B}, \text{C}, \text{N}$ ), which indicates that the Fe moments are sensitive to the local short-range order in the cementite-type crystals.

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