See discussions, stats, and author profiles for this publication at: https://www.researchgate.net/publication/282980699

# Structure Governs the Deformability of Polymer Particles in a Microfluidic Blood Capillary Model

ARTICLE in ACS MACRO LETTERS · OCTOBER 2015

Impact Factor: 5.76 · DOI: 10.1021/acsmacrolett.5b00591

CITATION **READS** 

1

48

# 8 AUTHORS, INCLUDING:



Huanli Sun

Soochow University (PRC)

18 PUBLICATIONS 650 CITATIONS

SEE PROFILE



Jiwei Cui

University of Melbourne

**56** PUBLICATIONS **900** CITATIONS

SEE PROFILE



Mattias Björnmalm

University of Melbourne

18 PUBLICATIONS 148 CITATIONS

SEE PROFILE



**Edgar Wong** 

University of New South Wales

41 PUBLICATIONS 434 CITATIONS

SEE PROFILE



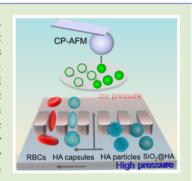
# Structure Governs the Deformability of Polymer Particles in a Microfluidic Blood Capillary Model

Huanli Sun,<sup>†,‡,§</sup> Mattias Björnmalm,<sup>†,‡,§</sup> Jiwei Cui,<sup>†,‡</sup> Edgar H. H. Wong,<sup>‡</sup> Yunlu Dai,<sup>†,‡</sup> Qiong Dai,<sup>†,‡</sup> Greg G. Qiao,\*<sup>,‡</sup> and Frank Caruso\*,<sup>†,‡</sup>

<sup>†</sup>ARC Centre of Excellence in Convergent Bio-Nano Science and Technology and <sup>‡</sup>Department of Chemical and Biomolecular Engineering, The University of Melbourne, Parkville, Victoria 3010, Australia

Supporting Information

ABSTRACT: Particle stiffness is a design parameter that affects bionano interactions, including biodistribution kinetics and cellular processing. Herein, we develop soft polysaccharide (hyaluronic acid, HA) replica particles and capsules with tunable stiffness and sizes similar to human red blood cells (RBCs) via atom transfer radical polymerization-mediated continuous assembly of polymers (CAP<sub>ATRP</sub>) and investigate their stiffness and deformability using colloidal-probe atomic force microscopy (CP-AFM) and a microfluidic blood capillary model, respectively. We demonstrate that HA replica particles and capsules with comparable nanoscale stiffness exhibit significantly different behaviors in a microfluidic blood capillary model. HA capsules behaved as RBCs, while HA replica particles had difficulty passing through the capillaries. These results (i) demonstrate how flow-based deformability measurements can be used to complement nanoscale stiffness measurements and (ii) provide important insight into the role of particle structure on the flow-based deformability of soft replica particles and capsules in a physiologically relevant microfluidic model.



E ngineered nanostructured particles are of interest due to their potential application in various fields, including drug delivery, nanoreactors, sensors, and artificial organelles. The performance of engineered particles in biological environments is governed by their ability to negotiate biological barriers, half which is determined by both the biological target environment and the physicochemical properties of the particles, including size, shape, surface chemistry, and stiffness.

Stiffness and deformability are particle characteristics that have recently gained increasing interest, as they have been shown to strongly influence the cellular interaction and processing of engineered particles. <sup>13–17</sup> An example of reversible deformability found in nature is that of human red blood cells (RBCs), which routinely and repeatedly deform to pass through blood capillaries smaller than themselves. 18,19 Inspired by this, several types of particles with varying stiffnesses and shapes have been engineered and subjected to constricted channels to investigate their deformability.<sup>20–27</sup> Soft particles with high deformability were found to deform to pass through constricted environments, a behavior that correlated with extended circulation times in vivo.<sup>28-30</sup> Although these studies provide important insights into how particles of various materials can be engineered to have different mechanical properties and biological behavior, the effect of the internal structure of soft polymer particles on deformability behavior remains to be explored.

Herein, we use a recently developed templating technique, called continuous assembly of polymers (CAP), 14,31-33 to engineer polymer capsules (i.e., with a hollow interior) and porous replica particles (i.e., with a structured interior) with

tunable stiffness and similar size to RBCs and investigate their deformability through liquid colloidal-probe atomic force microscopy (CP-AFM)<sup>34,35</sup> and a microfluidic blood capillary model (Scheme 1).<sup>27</sup> The replica particles and capsules were both made of hyaluronic acid (HA) and could be engineered with tunable stiffness, as assessed by CP-AFM. The flow-based deformability of the replica particles and capsules was then investigated using a microfluidic blood capillary model, and interestingly, a clear difference was observed between capsules and replica particles, even though both particle types had similar nanoscale stiffness. These results provide key insight into the importance of internal particle structure on the overall physical and mechanical properties and highlight how highresolution CP-AFM results can be complemented by largerscale flow-based deformability measurements to facilitate the rational design of soft particles with enhanced performance in physiological environments.

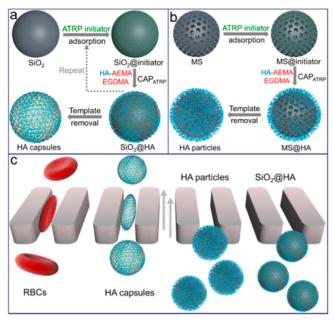
To prepare HA capsules and replica particles, atom transfer radical polymerization-mediated CAP (CAP<sub>ATRP</sub>) was used. HA, a natural polysaccharide, was chosen to construct the capsules and replica particles due to its high biocompatibility, nonimmunogenicity, and targeting ability, which are important for any future biomedical applications. HA capsules with similar size to RBCs were prepared via CAP<sub>ATRP</sub> of methacrylate-functionalized HA (HA-AEMA, see Scheme S1a in Supporting Information) on sacrificial SiO<sub>2</sub> particles (7.0  $\mu$ m

Received: August 20, 2015 Accepted: October 12, 2015



ACS Macro Letters

Scheme 1. Schematic Illustration of  $CAP_{ATRP}$ -Mediated Formation of (a) HA Hollow Capsules Using  $SiO_2$  Templates, and (b) HA Replica Particles Using Mesoporous Silica (MS) Templates and (c) Their Deformability in a Microfluidic Blood Capillary Model<sup>a</sup>



"HA-coated SiO<sub>2</sub> particles (SiO<sub>2</sub>@HA) and RBCs were used as controls. Full experimental details are available in the Supporting Information.

in diameter) and subsequent template removal. The CAP<sub>ATRP</sub> step number was altered from 1 to 4 to afford HA capsules (HA<sub>C1</sub>, HA<sub>C2</sub>, HA<sub>C3</sub>, and HA<sub>C4</sub>, respectively) with different wall thickness and stiffness, as reported previously.<sup>14</sup> HA capsules were further labeled with an Alexa Fluor 488 *N*-hydroxysuccinimide ester to enable fluorescence imaging. The capsules were imaged using fluorescence, differential interference contrast (DIC), and deconvolution microscopy, as well as transmission

electron microscopy (TEM) and AFM (Figure 1a1–e1 and Figure S1). The HA capsules were well dispersed in aqueous solution with a hollow structure and a diameter of  $\sim 7~\mu m$ . All HA capsules (HA<sub>C1</sub>–HA<sub>C4</sub>) showed a collapsed structure with folds and creases in the air-dried state, as commonly observed for hollow capsules.  $^{38}$  In addition, AFM analysis of HA capsules showed a linear increase in capsule wall thickness from  $\sim \! 4.5$  nm (HA<sub>C1</sub>) to  $\sim \! 15.9$  nm (HA<sub>C4</sub>, Figure S2) with the number of CAP<sub>ATRP</sub> steps, which is consistent with our previous study of HA film growth on 2.4  $\mu m$  SiO<sub>2</sub> particles.  $^{14}$ 

For construction of HA replica particles, one CAP<sub>ATRP</sub> step of HA-AEMA was performed on mesoporous silica (MS) templates (average diameter  $\sim 7.2 \mu m$ ). MS particles were functionalized with an ATRP macroinitiator (Scheme S1b) and subsequently incubated with HA-AEMA and ethylene glycol dimethacrylate (EGDMA) solution for 2 h prior to CAP<sub>ATRP</sub> by mixing with PMDETA, NaAsc, and CuBr<sub>2</sub> (Scheme 1b). Various amounts of HA-AEMA (HA to MS weight ratio of 10%, 20%, 40%, and 80%) were used to construct replica particles with different HA densities upon removal of MS templates, yielding HA<sub>P1</sub>, HA<sub>P2</sub>, HA<sub>P3</sub>, and HA<sub>P4</sub>, respectively. The obtained replica particles were characterized, as shown in Figure 1a2-e2 and Figure S3. The HA replica particles were well dispersed in aqueous solution and had a size of  $\sim 6-8 \mu m$ . However, these particles had a different morphology after airdrying, compared with the HA capsules (Figure 1c1): unlike HA capsules, which showed a fully collapsed structure, HA replica particles maintained more of their spherical shape. This has been observed for other polymer replica particles constructed by surface-initiated ATRP<sup>39</sup> or through a polymer infiltration-cross-linking method. 40 AFM analysis also showed that the height (thickness) of air-dried HA replica particles increased, from  $\sim$ 690 nm (HA<sub>P1</sub>) to  $\sim$ 1200 nm (HA<sub>P4</sub>), as the initial HA to MS weight ratio increased (Figure S2b), indicating a higher polymer density in HA replica particles prepared using a larger amount of HA-AEMA. Deconvolution microscopy images confirmed the hollow interior of the capsules and a filled interior of the replica particles, demonstrating the difference in internal structure (Figure 1d,e and Figure S4).

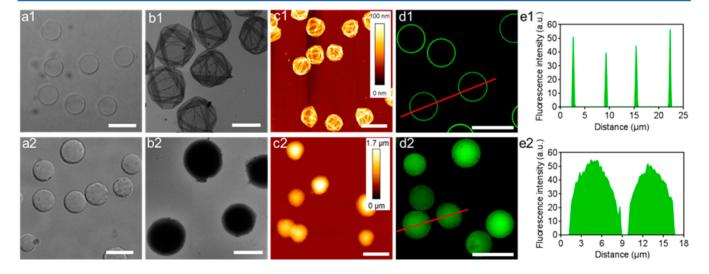
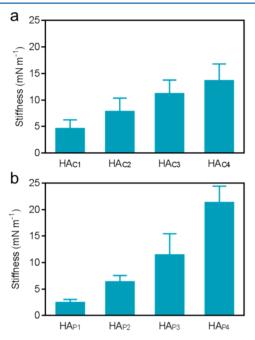


Figure 1. Structural characterization of  $HA_{C3}$  capsules (a1–e1) and  $HA_{P3}$  replica particles (a2–e2). (a) DIC microscopy, (b) TEM, (c) AFM, and (d) fluorescence deconvolution microscopy (single z-plane) of  $HA_{C3}$  capsules and  $HA_{P3}$  replica particles, respectively. (e) The fluorescence intensity profiles of the red line marked in d1 and d2, respectively. a1, a2, d1, and d2 are capsules/particles in solution. b1 and b2 are capsules/particles under vacuum, and c1 and c2 are capsules/particles in the air-dried state. Scale bars are 10  $\mu$ m in a, c, and d and 5  $\mu$ m in b.

ACS Macro Letters Letter

For particles with the same material composition, the material density and/or amount can be used to tune stiffness.  $^{15,41-43}$  In the current study, the amount of HA-AEMA used during CAP<sub>ATRP</sub> was varied, and high-resolution nanometer-scale deformation measurements (100 nm deformation) were made using CP-AFM on individual capsules and replica particles in phosphate-buffered saline (PBS) solution.  $^{44,45}$  It was found that capsule stiffness increased almost linearly with increasing HA layer number, with the measured stiffness of capsules ranging from  $\sim$ 4.6 (HA<sub>C1</sub>) to 13.6 mN m<sup>-1</sup> (HA<sub>C4</sub>, Figure 2a). The stiffness of HA replica particles could



**Figure 2.** Stiffness of (a) HA capsules with a different number of HA layers and (b) HA replica particles prepared using different HA-AEMA to MS weight ratios, as determined by CP-AFM analysis.

also be finely tuned by changing the concentration of HA-AEMA in the CAP<sub>ATRP</sub> step, with values ranging from  $\sim$ 2.4 (HA<sub>P1</sub>) to 21.3 mN m<sup>-1</sup> (HA<sub>P4</sub>, Figure 2b). The observed trend is consistent with a previous report where an increase in cross-linker amount led to an increase in particle stiffness. Moreover, the stiffness range measured for the replica particles was similar to that for the HA capsules.

The flow-based deformability of the particles was investigated in a microfluidic blood capillary model (Figure 3). HA<sub>C3</sub> and HA<sub>P3</sub> had similar nanoscale stiffness (11.1 and 11.4 mN m<sup>-1</sup>, respectively) and were therefore compared using a recently introduced microfluidic blood capillary model.<sup>27</sup> This model was designed using physiologically relevant dimensions and pressure drops across capillaries. The capsules and replica particles were injected from a high-pressure inlet, and only buffer was injected from a low-pressure inlet. The four-point device design facilitates the establishment of well-controlled physiologically relevant differential pressures and pressure drops across the capillaries, as each inlet can be tuned independently (Figure 3a). The HA<sub>C3</sub> capsules could deform and passed through the capillaries (Figure 3b) with no visibly damaged capsules seen in either of the outlets, thus indicating that the capsules could pass through the capillaries via reversible elastic deformation (Figure 3c). However, a different

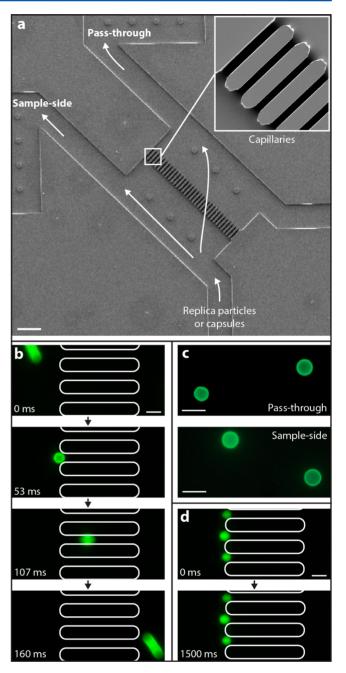


Figure 3. (a) Illustration of available flow paths for HA capsules and replica particles in the microfluidic blood capillary model, overlaid on a SEM image of the PDMS microfluidic device. (b) Time-lapse fluorescence microscopy images of a HA<sub>C3</sub> capsule passing through a capillary. (c) Fluorescence microscopy image of capsules at the outlets of the device after passing through capillaries (pass-through) or staying on the same side (sample side). Note that the capsules were not visibly damaged. (d) Time-lapse fluorescence microscopy images of HA<sub>P3</sub> replica particles at the entrance of capillaries at lower capillary pressure drops. The pressure drop across the capillaries is  $\sim\!\!6$  mbar in b and d. The nonspherical ("drawn out") shape of the capsules observed in some of the time-lapse frames is due to the acquisition rate of the camera. Spherical capsules are again observed when the capsules are no longer in motion (as seen in (c)). Scale bars are 100  $\mu$ m in a and 10  $\mu$ m in b–d.

behavior was observed for the  $HA_{P3}$  replica particles, as they had difficulty passing through the capillaries, especially at lower pressure differentials.

ACS Macro Letters Letter

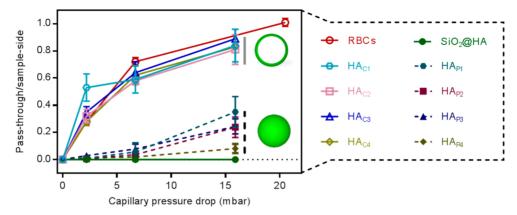


Figure 4. Particle trajectory analysis of HA capsules and replica particles with varying stiffness.  $SiO_2@HA$  and RBCs were used as controls. Number of capsules or replica particles that passed through the capillaries was divided by the number that went to the outlet on the sample side. Capsule data points are indicated by a hollow green circle, and replica particle data points are indicated by a solid green sphere. Mean  $\pm$  standard deviation, n = 3.

To quantify the differences found in the flow-based deformability behavior of the different particles, particle trajectories in the device were analyzed, and a ratio of the number of particles that passed through the capillaries to the number of particles that stayed on the sample side was calculated (Figure 4). Particles that can easily pass through will have a high ratio, while particles that do not pass through will have a ratio of zero. Each measurement was performed in triplicate with more than 100 trajectories analyzed for each replicate. In total, over 10 000 particle trajectories were analyzed. The controls behaved as expected, with RBCs easily passing through the capillaries and the ratio increasing with increasing pressure drop, while none of the SiO<sub>2</sub>@HA particles (i.e., with core remaining) could pass through, resulting in a ratio of zero, even at higher pressure drops. The capsules (HA<sub>C1</sub>-HA<sub>C4</sub>) could all pass through the capillaries with relative ease and exhibited deformation-based trajectories in the microfluidic blood capillary similar to the RBCs. However, the replica particles (HA<sub>P1</sub>-HA<sub>P4</sub>) had difficulty passing through the capillaries, even at higher pressure drops. Even when comparing the capsule with the highest nanoscale stiffness (HA<sub>C4</sub>, 13.6 mN m<sup>-1</sup>) to the replica particle with the lowest nanoscale stiffness (HA<sub>P1</sub>, 2.4 mN m<sup>-1</sup>), it was observed that the capsules had a significantly higher ratio, despite the fact that the capsules were almost 6-fold higher in nanoscale stiffness. As the main difference between the two particle types is their internal structure, where both particle types were made of the same materials and engineered to have the same stiffness at the nanoscale, this demonstrates the importance of particle structure on flow-based deformability in the microfluidic blood capillary model. These results also show the importance of investigating stiffness at different length scales, both at the nanoscale (e.g., using CP-AFM) and at the microscale (e.g., using microfluidic models), to obtain a more complete understanding of the deformability of soft microparticles.

In summary, HA replica particles and HA capsules were nanoengineered through CAP<sub>ATRP</sub>. Both particle types were well dispersed in aqueous solution and were of similar size to each other and to human RBCs. The nanoscale stiffness of both particle types was investigated using CP-AFM, and it was demonstrated that both replica particles and capsules could be engineered to have similar nanoscale stiffness, with capsule stiffness ranging from 4.6 to 13.6 mN m<sup>-1</sup> and replica particle stiffness from 2.4 to 21.3 mN m<sup>-1</sup>. A microfluidic blood capillary model was then used to investigate the behavior of

both particle types in a physiologically relevant flow environment. It was found that the flow-based deformability behavior and the ability to pass through the capillaries for capsules were similar to those of RBCs, while replica particles could not easily pass through the capillaries, even at higher capillary pressure drops. These results highlight how flow-based deformability measurements can complement nanoscale stiffness measurements and demonstrate the impact of particle structure on the overall deformability behavior of particles, thus underlining particle structure as an important parameter that, together with particle stiffness, has the potential for fine-tuning particle behavior in flow-based environments, both in vitro and in vivo. These results can therefore facilitate the rational design of particle systems with improved ability to negotiate biological barriers at multiple length scales, which is important for advancing the use of engineered particles in biomedical applications such as imaging, diagnostics, and drug delivery.

# ASSOCIATED CONTENT

#### S Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsmacrolett.5b00591.

Full experimental details, supporting scheme, and supporting figures (PDF)

#### AUTHOR INFORMATION

# **Corresponding Authors**

\*E-mail: gregghq@unimelb.edu.au (G.G.Q.). \*E-mail: fcaruso@unimelb.edu.au (F.C.).

# **Author Contributions**

§These authors contributed equally. The manuscript was written through contributions of all authors.

#### Note

The authors declare no competing financial interest.

# ACKNOWLEDGMENTS

This research was conducted and funded by the Australian Research Council (ARC) Centre of Excellence in Convergent Bio-Nano Science and Technology (CE140100036). This work was also supported by the ARC under the Australian Laureate Fellowship (FL120100030, F.C.), Future Fellowship (FT110100411, G.G.Q.), Discovery Project (DP1094147 and DP130101846, F.C. and G.G.Q.), and Super Science Fellowship

ACS Macro Letters Letter

ship (FS110200025, F.C. and G.G.Q.) schemes, as well as the Australian Government through an Australian Postgraduate Award (M.B.). We acknowledge Martin P. van Koeverden for providing the ATRP macroinitiator and Junling Guo, Katelyn T. Gause, and Dr. Biao Kong for helpful discussions. This work was performed in part at the Materials Characterisation and Fabrication Platform (MCFP) at The University of Melbourne and the Victorian Node of the Australian National Fabrication Facility (ANFF).

#### REFERENCES

- (1) De Koker, S.; Hoogenboom, R.; De Geest, B. G. Chem. Soc. Rev. **2012**, 41, 2867–2884.
- (2) Tanner, P.; Baumann, P.; Enea, R.; Onaca, O.; Palivan, C.; Meier, W. Acc. Chem. Res. **2011**, 44, 1039–1049.
- (3) Gokmen, M. T.; Du Prez, F. E. Prog. Polym. Sci. 2012, 37, 365-405.
- (4) Yan, Y.; Björnmalm, M.; Caruso, F. ACS Nano 2013, 7, 9512-9517.
- (5) Nicolas, J.; Mura, S.; Brambilla, D.; Mackiewicz, N.; Couvreur, P. *Chem. Soc. Rev.* **2013**, 42, 1147–1235.
- (6) Ernsting, M. J.; Murakami, M.; Roy, A.; Li, S. D. J. Controlled Release 2013, 172, 782-794.
- (7) Chauhan, V. P.; Jain, R. K. Nat. Mater. 2013, 12, 958-962.
- (8) Yan, Y.; Björnmalm, M.; Caruso, F. Chem. Mater. 2014, 26, 452–460.
- (9) Nel, A. E.; Madler, L.; Velegol, D.; Xia, T.; Hoek, E. M. V.; Somasundaran, P.; Klaessig, F.; Castranova, V.; Thompson, M. *Nat. Mater.* **2009**, *8*, 543–557.
- (10) Zhu, M.; Nie, G.; Meng, H.; Xia, T.; Nel, A.; Zhao, Y. Acc. Chem. Res. 2013, 46, 622–631.
- (11) Albanese, A.; Tang, P. S.; Chan, W. C. W. Annu. Rev. Biomed. Eng. 2012, 14, 1–16.
- (12) Petros, R. A.; DeSimone, J. M. Nat. Rev. Drug Discovery 2010, 9, 615–627.
- (13) Sun, J. S.; Zhang, L.; Wang, J. L.; Feng, Q.; Liu, D. B.; Yin, Q. F.; Xu, D. Y.; Wei, Y. J.; Ding, B. Q.; Shi, X. H.; Jiang, X. Y. *Adv. Mater.* **2015**, *27*, 1402–1407.
- (14) Sun, H.; Wong, E. H. H.; Yan, Y.; Cui, J.; Dai, Q.; Guo, J.; Qiao, G. G.; Caruso, F. Chem. Sci. 2015, 6, 3505–3514.
- (15) Hartmann, R.; Weidenbach, M.; Neubauer, M.; Fery, A.; Parak, W. J. Angew. Chem., Int. Ed. 2015, 54, 1365–1368.
- (16) Banquy, X.; Suarez, F.; Argaw, A.; Rabanel, J. M.; Grutter, P.; Bouchard, J. F.; Hildgen, P.; Giasson, S. *Soft Matter* **2009**, *5*, 3984–3991.
- (17) Best, J. P.; Yan, Y.; Caruso, F. Adv. Healthcare Mater. 2012, 1, 35–47.
- (18) Skalak, R.; Branemark, P. I. Science 1969, 164, 717-719.
- (19) Mohandas, N.; Gallagher, P. G. Blood 2008, 112, 3939-3948.
- (20) Björnmalm, M.; Yan, Y.; Caruso, F. J. Controlled Release 2014, 190, 139–149.
- (21) Doshi, N.; Zahr, A. S.; Bhaskar, S.; Lahann, J.; Mitragotri, S. *Proc. Natl. Acad. Sci. U. S. A.* **2009**, *106*, 21495–21499.
- (22) Haghgooie, R.; Toner, M.; Doyle, P. S. Macromol. Rapid Commun. 2010, 31, 128-134.
- (23) Hayashi, K.; Ono, K.; Suzuki, H.; Sawada, M.; Moriya, M.; Sakamoto, W.; Yogo, T. *Small* **2010**, *6*, 2384–2391.
- (24) She, S.; Xu, C.; Yin, X.; Tong, W.; Gao, C. Langmuir 2012, 28, 5010–5016.
- (25) Chen, K.; Merkel, T. J.; Pandya, A.; Napier, M. E.; Luft, J. C.; Daniel, W.; Sheiko, S.; DeSimone, J. M. *Biomacromolecules* **2012**, *13*, 2748–2759.
- (26) She, S. P.; Li, Q. Q.; Shan, B. W.; Tong, W. J.; Gao, C. Y. Adv. Mater. **2013**, 25, 5814–5818.
- (27) Cui, J.; Björnmalm, M.; Liang, K.; Xu, C.; Best, J. P.; Zhang, X.; Caruso, F. *Adv. Mater.* **2014**, *26*, 7295–7299.
- (28) Merkel, T. J.; Jones, S. W.; Herlihy, K. P.; Kersey, F. R.; Shields, A. R.; Napier, M.; Luft, J. C.; Wu, H.; Zamboni, W. C.; Wang, A. Z.;

Bear, J. E.; DeSimone, J. M. Proc. Natl. Acad. Sci. U. S. A. 2011, 108, 586–591.

- (29) Zhang, L.; Cao, Z.; Li, Y.; Ella-Menye, J.-R.; Bai, T.; Jiang, S. ACS Nano 2012, 6, 6681–6686.
- (30) Anselmo, A. C.; Zhang, M.; Kumar, S.; Vogus, D. R.; Menegatti, S.; Helgeson, M. E.; Mitragotri, S. *ACS Nano* **2015**, *9*, 3169–3177.
- (31) Goh, T. K.; Guntari, S. N.; Ochs, C. J.; Blencowe, A.; Mertz, D.; Connal, L. A.; Such, G. K.; Qiao, G. G.; Caruso, F. *Small* **2011**, *7*, 2863–2867.
- (32) Mertz, D.; Ochs, C. J.; Zhu, Z.; Lee, L.; Guntari, S. N.; Such, G. K.; Goh, T. K.; Connal, L. A.; Blencowe, A.; Qiao, G. G.; Caruso, F. *Chem. Commun.* **2011**, 47, 12601–12603.
- (33) Wong, E. H. H.; Guntari, S. N.; Blencowe, A.; van Koeverden, M. P.; Caruso, F.; Qiao, G. G. ACS Macro Lett. 2012, 1, 1020–1023.
- (34) Ferv. A.; Dubreuil, F.; Möhwald. H. *New J. Phys.* **2004**, *6*, 18.
- (35) Vinogradova, O. I.; Lebedeva, O. V.; Kim, B. S. Annu. Rev. Mater. Res. 2006, 36, 143–178.
- (36) Kogan, G.; Soltes, L.; Stern, R.; Gemeiner, P. Biotechnol. Lett. **2006**, 29, 17–25.
- (37) Ossipov, D. A. Expert Opin. Drug Delivery 2010, 7, 681-703.
- (38) Donath, E.; Sukhorukov, G. B.; Caruso, F.; Davis, S. A.; Möhwald, H. *Angew. Chem., Int. Ed.* **1998**, *37*, 2201–2205.
- (39) Müllner, M.; Cui, J.; Noi, K. F.; Gunawan, S. T.; Caruso, F. Langmuir **2014**, 30, 6286–6293.
- (40) Cui, J.; Yan, Y.; Wang, Y.; Caruso, F. Adv. Funct. Mater. 2012, 22, 4718–4723.
- (41) Palankar, R.; Pinchasik, B.-E.; Schmidt, S.; De Geest, B. G.; Fery, A.; Möhwald, H.; Skirtach, A. G.; Delcea, M. *J. Mater. Chem. B* **2013**, *1*, 1175–1181.
- (42) Delcea, M.; Schmidt, S.; Palankar, R.; Fernandes, P. A. L.; Fery, A.; Möhwald, H.; Skirtach, A. G. *Small* **2010**, *6*, 2858–2862.
- (43) Cui, J.; De Rose, R.; Best, J. P.; Johnston, A. P. R.; Alcantara, S.; Liang, K.; Such, G. K.; Kent, S. J.; Caruso, F. *Adv. Mater.* **2013**, 25, 3468–3472.
- (44) Fery, A.; Weinkamer, R. Polymer 2007, 48, 7221-7235.
- (45) Schmidt, S.; Fernandes, P. A. L.; De Geest, B. G.; Delcea, M.; Skirtach, A. G.; Möhwald, H.; Fery, A. *Adv. Funct. Mater.* **2011**, *21*, 1411–1418.