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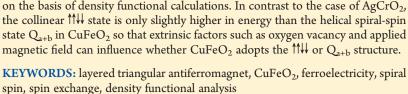
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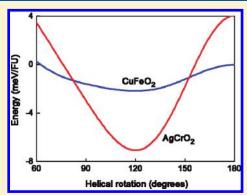


Density Functional Investigation of the Difference in the Magnetic Structures of the Layered Triangular Antiferromagnets CuFeO₂ and AgCrO₂

Yuemei Zhang, [†] Erjun Kan, [‡] and Myung-Hwan Whangbo*, [†]

ABSTRACT: The layered triangular antiferromagnets CuFeO₂ and AgCrO₂ exhibit ferroelectric polarization when they adopt a helical spiral-spin magnetic structure. Application of magnetic field parallel to the *c*-axis or doping Fe sites with Al or Ga is necessary for CuFeO₂ to have a helical spiral spin order. However, AgCrO₂ adopts a helical spiral spin order in the absence of such extrinsic factors. A probable cause for this difference between the two systems was examined by evaluating the relative stabilities of their helical spiral and collinear spin structures on the basis of density functional calculations. In contrast to the case of AgCrO₂, the collinear \frac{\partitus}{\partitus} state is only slightly higher in energy than the helical spiral-spin state Q_{a+b} in CuFeO₂ so that extrinsic factors such as oxygen vacancy and applied





1. INTRODUCTION

Multiferroic (i.e., magnetic ferroelectric) materials have received much attention due to the potential of controlling their physical properties by either magnetic or electric field. A centrosymmetric crystalline solid can become ferroelectric (FE) when it loses inversion symmetry, which is often induced either by cooperative Jahn-Teller distortion 1b,2 or by chiral magnetic order. 1f,3 The FE polarization driven by magnetic order has been discussed in terms of exchange striction⁴ or spin-orbit coupling (SOC).⁵ The layered triangular antiferromagnets such as CuFeO₂⁶ and AgCrO₂⁷ exhibit FE polarization when their magnetic structure acquires a helical spiral-spin order in the chains of M^{3+} (M = Fe, Cr) ions of their MO₂ layers. The MO₂ layers of AMO₂ (A = Cu, M = Fe; A = Ag, M = Cr) are made up of edge-sharing MO₆ octahedra ^{8,9} containing high-spin M³⁺ (d⁵, S = 5/2 for M = Fe; d^3 , S = 3/2 for M = Cr) ions (Figure 1a). ¹⁰ Each MO₂ layer can be viewed in terms of the MO₄ chains of edge-sharing FeO₆ octahedra running along the a-, b-, or (a+b)direction. Adjacent MO₂ layers of AMO₂ are stacked along the c-direction with A+ (d10) ions intercalated between the MO2 layers to form linear O-A-O bridges (Figure 1b) such that there are three MO₂ layers per unit cell. Although CuFeO₂ and AgCrO₂ are similar in crystal structure, they show an interesting difference in magnetic structure. The magnetic ground state of CuFeO₂ has the $\uparrow\uparrow\downarrow\downarrow$ spin arrangement in each FeO₂ layer (Figure 2a) with their spins parallel to the *c*-axis. ¹¹ This magnetic structure is modified by magnetic field H applied along the c-direction; the magnetic structure changes from the 1114 arrangement under $H < \sim 7$ T to the helical spiral-spin structure under $H \approx 7-13$ T to the $\uparrow \uparrow \uparrow \downarrow \downarrow$ structure (Figure 2b) under

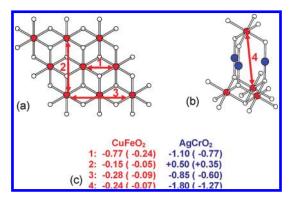


Figure 1. (a) An isolated FeO₂ layer of CuFeO₂ made of edge-sharing FeO₆ octahedra showing the intralayer spin exchange paths J_1 , J_2 , and J_3 . (b) A zoomed-in view of how two adjacent FeO₂ layers of CuFeO₂ are linked by O-Cu-O bridges showing the interlayer spin exchange J₄. The Fe, Cu, and O atoms are indicated by red, blue, and white circles, respectively. The numbers 1-4 in (a, b) refer to the spin exchange paths J_1-J_4 , respectively. (c) The values of J_1-J_4 (in meV) found for CuFeO₂ and AgCrO₂ obtained from GGA+U calculations. The numbers outside the parentheses are the as-calculated values, and those inside the parentheses are the reduced values using the overestimation factor f = 3.24 for CuFeO₂ and f = 1.42 for AgCrO₂ (see the text for details).

 $H > \sim 13 \text{ T}$. The helical spiral-spin state can also be induced into CuFeO2 without applying magnetic field by doping the Fe

Received: April 22, 2011 August 24, 2011 Revised:

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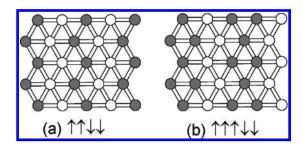


Figure 2. The collinear magnetic structures of $CuFeO_2$ below 10 K under magnetic field: (a) the $\uparrow\uparrow\downarrow\downarrow$ structure under magnetic field lower than \sim 7 T and (b) the $\uparrow\uparrow\uparrow\downarrow\downarrow$ structure under magnetic field greater than \sim 13 T. The filled and empty circles represent the spins of the Fe^{3+} ions that are aligned along the positive and negative c-axis directions, respectively.

sites with Al or Ga atoms. ^{6b,13} In contrast, AgCrO₂ adopts a helical spiral-spin state in the absence of applied magnetic field H.

The magnetic structure of CuFeO₂ at low temperature was examined by inelastic neutron scattering measurements, 14 and the resulting spin wave dispersion relations were theoretically analyzed to find that the three intralayer spin exchanges $J_1 - J_3$ and the interlayer spin exchange J₄ (Figure 1) are all antiferromagnetic. Other magnetic properties of CuFeO2 were also investigated in a number of studies. 16 So far, however, it has not been explained why CuFeO2 requires a magnetic field applied parallel to the c-axis to adopt a helical spiral spin order, but AgCrO₂ does not. To answer this question, one needs to estimate the relative energies of the collinear and helical spiral spin states of CuFeO2 and AgCrO2, which in turn requires the knowledge of their spin exchange interactions. The spin exchange parameters of CuFeO₂ were deduced experimentally, 14,15 but those of AgCrO₂ were not. In contrast, the spin exchange parameters of AgCrO2 were evaluated on the basis of density functional theory (DFT) calculations, 7b but those of CuFeO₂ were not.

In the present work, we examine the difference in the magnetic structures of $CuFeO_2$ and $AgCrO_2$ on the basis of DFT calculations. We first evaluate the spin exchange parameters of $CuFeO_2$ by performing DFT calculations and then determine the relative stabilities of the helical spiral and collinear spin structures of $CuFeO_2$ and $AgCrO_2$ in terms of their spin exchange parameters and finally propose a most-likely reason why $CuFeO_2$ and $AgCrO_2$ differ in their magnetic structures under magnetic field.

2. DETAILS OF CALCULATIONS

Our spin-polarized DFT calculations employed the projector augmented wave method 17 coded in the Vienna ab initio simulation package, 18 the generalized gradient approximation (GGA) for the exchange and correlation corrections, 19 and the plane wave cutoff energies of 500 and 400 eV for the calculations without and with including the effect of SOC, respectively. To describe the electron correlation associated with the Fe 3d states, the GGA plus on-site repulsion U (GGA+U) method was applied with an effective U = 4 eV on the Fe atom. As in the case of ${\rm AgCrO_{22}}^{7b}$ we consider the four spin exchange interactions ${\rm J_1-J_4}$ defined in Figure 1 and evaluate their values by performing GGA+U calculations for the five ordered spin states of CuFeO₂ defined in Figure 3.

3. SPIN EXCHANGE INTERACTIONS

We determine the four spin exchange parameters J_1-J_4 of CuFeO₂ (Figure 1) by performing mapping analysis²⁰ on the

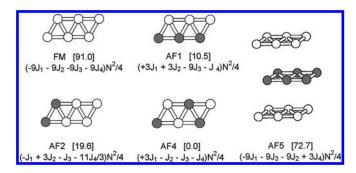


Figure 3. The five ordered spin states of CuFeO₂ constructed using a (3a, 2b, c) supercell, where the filled and empty circles represent up-spin and down-spin Fe³⁺ sites, respectively. In (a–d), all three FeO₂ layers of the (3a, 2b, c) supercell have the same spin arrangement, so that the spin arrangement of only one FeO₂ layer is shown. In (e), the spins are ferromagnetic in each FeO₂ layer, but the (3a, 2b, c) supercell consists of one up-spin and two down-spin layers. For each state, the number in each square bracket refers to the relative energy per three FUs determined from GGA+U calculations, and the energy expression to the total spin exchange energy per three FUs with N=5.

basis of the five ordered spin states (FM, AF1, AF2, AF3, AF4) shown in Figure 3. The relative energies, per three formula units (FUs), of these states were determined from our GGA+U calculations as summarized in Figure 3. The energies of these states can also be described in terms of the spin Hamiltonian,

$$\hat{H} = -\sum_{i < j} J_{ij} \hat{S}_i \cdot \hat{S}_j \tag{1}$$

where J_{ij} (= J_1 – J_4) is the spin exchange parameter for the spin exchange interaction between the spin sites i and j. By applying the energy expressions obtained for spin dimers with N unpaired spins per spin site (N = 5 for CuFeO₂), ²⁰ the total spin exchange energies of the five ordered spin states (per three FUs) are written as summarized in Figure 3. Thus, by mapping the relative energies of the five ordered spin states determined by the GGA +U calculations onto the corresponding relative energies determined from the above spin exchange energies, we obtain the values of J_1 – J_4 summarized in Figure 1c, where the corresponding values of $AgCrO_2$ are also listed for comparison. (To compare the spin exchange interactions of $CuFeO_2$ with those of $AgCrO_2$, it is necessary to use the effective spin exchanges $J_i^{eff} = S^2J_{ii}$ where S = 5/2 and 3/2 for $CuFeO_2$ and $AgCrO_2$, respectively.)

Figure 1c shows that the intralayer exchanges J_1 and J_3 as well as the interlayer exchange J₄ of CuFeO₂ are substantially antiferromagnetic (AFM), namely, $J_1-J_4 = -0.77$, -0.15, -0.28, and -0.24 meV, respectively. These values are the same in sign as, and are comparable in magnitude to, the corresponding spin exchanges deduced from the inelastic neutron scattering measurements (namely, -0.45, -0.20, -0.26, and -0.13 meV, respectively). 14 In each FeO2 layer, spin frustration 21 occurs in the (J_1, J_1, J_1) and (J_3, J_3, J_3) triangles and in the (J_1, J_1, J_3) line segments. Between adjacent FeO2 layers, spin frustration occurs in the isosceles (J_1, J_4, J_4) triangles. That is, both the intralayer and the interlayer interactions are spin-frustrated in CuFeO₂, as found for AgCrO₂. 7b The spin exchange interactions of CuFeO₂ differ from those of AgCrO₂ in two aspects: (a) Among J₁-J₄, the AFM interlayer spin exchange J₄ is the strongest interaction in AgCrO₂ but is quite weak in CuFeO₂, and (b) the intralayer exchange J2 is ferromagnetic (FM) in AgCrO2 but is AFM in

CuFeO₂. As discussed in the next section, these two differences are important in explaining why a magnetic field applied parallel to the *c*-axis is necessary for CuFeO₂ to adopt the helical spiral-spin structure while this is not the case for AgCrO₂.

It is known that GGA+U electronic structure calculations generally overestimate the magnitude of spin exchange interactions. To find the overestimation factors f for the calculated spin exchange parameters of CuFeO₂ and AgCrO₂ (Figure 1c), we calculate their Curie—Weiss temperatures θ using the mean field approximation, ²³ namely,

$$\theta = \frac{S(S+1)}{3k_B} \sum_{i} z_{i} J_{i}$$
 (2)

where the summation runs over all nearest neighbors of a given spin site, z_i is the number of nearest neighbors connected by the spin exchange parameter J_{ij} and S is the spin quantum number of each spin site (i.e., S = 5/2 for Fe^{3+} in $CuFeO_2$ and S = 3/2 for Cr^{3+} in $AgCrO_2$). Thus, for $CuFeO_2$ and $AgCrO_2$, θ can be approximated by

$$\theta \approx 2S(S + 1)(J_1 + J_2 + J_3 + J_4)/k_B$$
 (3)

The as-calculated J_1-J_4 values of Figure 1c lead to $\theta=-292$ and -283 K for CuFeO₂ and AgCrO₂, respectively, while the corresponding experimental values are -90^{24} and -200 K, 25 respectively. Thus, according to the experimental Curie—Weiss temperatures and the mean-field theory, the as-calculated J_1-J_4 values are overestimated by a factor of f=3.24 and 1.42 for CuFeO₂ and AgCrO₂, respectively. Figure 1c also lists, in parentheses, the values of J_1-J_4 reduced using the overestimation factor f. For CuFeO₂, the reduced J_1-J_4 values are -0.24, -0.05, -0.09, and -0.07 meV, respectively, which are smaller in magnitude than the corresponding values deduced from the inelastic neutron scattering measurements 14 (see above) approximately by a factor of 2.

4. RELATIVE STABILITIES OF THE HELICAL SPIRAL AND COLLINEAR SPIN STRUCTURES

To probe why a magnetic field applied parallel to the c-axis is needed for CuFeO₂ to acquire a helical spiral-spin structure, we first determine the magnetic anisotropy of CuFeO₂ by performing GGA+U+SOC calculations for its $\uparrow\uparrow\downarrow\downarrow$ state. These calculations reveal that the spin orientation parallel to the c-axis (//c) is more stable than that perpendicular to the c-axis ($\perp c$) by 0.12 meV per FU; namely, the magnetic anisotropy of each Fe³⁺ (S = 5/2) ion is very weak. This is not surprising because, to a first approximation, L = 0 for the Fe³⁺ ion in each FeO₆ octahedron so that the effect of SOC should be weak. This suggests that the spin orientation of CuFeO₂ might be easily influenced by extrinsic factors such as oxygen vacancy and applied magnetic field.

Since $CuFeO_2$ can switch from the $\uparrow\uparrow\downarrow\downarrow$ state to the helical spiral-spin state depending on the strength of applied magnetic field $\mathbf{H} < \sim 13$ T, the two states should be close in energy. (Note that 1 T is equivalent to 0.67 K.) Such a field-dependence is not known for $AgCrO_2$, which adopts the helical spiral-spin state, thereby implying that the $\uparrow\uparrow\downarrow\downarrow$ state is much less stable than the helical spiral-spin state in $AgCrO_2$. To verify these implications, we examine the relative stabilities of several helical spiral and collinear spin states of $CuFeO_2$ and $AgCrO_2$ in terms of their spin exchange parameters J_1-J_4 . In estimating the relative stabilities of these states, we employ the reduced J_1-J_4 values obtained from the as-calculated values by the overestimation

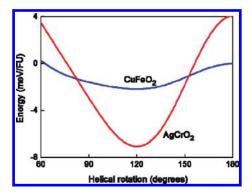


Figure 4. Plots of the spin exchange energy $E(\theta)$, per FU, of the helical spiral-spin structure $Q_{a+b} = (q, q)$ calculated as a function of the helical rotation angle $\theta = q \times 360^{\circ}$ for CuFeO₂ and AgCrO₂ using their spin exchange parameters $J_1 - J_4$ reduced by the overestimation factor f = 3.24 for CuFeO₂ and by f = 1.42 for AgCrO₂.

factor f. However, the use of the as-calculated J_1-J_4 values leads essentially to the same conclusion. As a representative example of the helical spiral-spin structure, we consider the helical spiral-spin structure $Q_{a+b}=(q,q)$ as a function of the helical rotation angle $\theta=q\times360^\circ$ (q=1/5,1/4,1/3,1/2). In $Q_{a+b}=(1/3,1/3)$, for example, the helical spiral-spin chains run along the positive a-direction with clockwise helical-rotation of 120° , and the spiral-spin chains repeat along the positive b-direction while advancing the helical-rotation clockwise by 120° . In terms of J_1-J_4 , the energy $E(\theta)$ of the Q_{a+b} structure as a function of θ is written as

$$E(\theta) = \{ -3J_1[2\cos\theta + \cos 2\theta] - 3J_2[1 + 2\cos 3\theta] - 3J_3[2\cos 2\theta + \cos 4\theta] - J_4[3 + 4\cos\theta + 2\cos 2\theta] \} (N^2/4)$$
(4)

The $E(\theta)$ vs θ plots calculated for CuFeO₂ and AgCrO₂ using their spin exchange parameters (reduced by the overestimation factor f) are presented in Figure 4 in the region of $\theta = 60 - 180^{\circ}$ (i.e., q = 1/6 - 1/2).

$$\Delta E = (-0.50J_1 + 2J_2 - 0.50J_3 - 0.33J_4)(N^2/4)$$
 (5)

where N=5 for CuFeO₂ and N=3 for AgCrO₂. Thus, ΔE is large for AgCrO₂ due to the strong AFM J₄ and the FM J₂, while ΔE is small for CuFeO₂ due to the weak AFM J₄ and the AFM J₂.

Since the Q_{a+b} (q=1/3) and $\uparrow\uparrow\downarrow\downarrow$ states of CuFeO₂ are close in energy, the extrinsic factors such as oxygen vacancy and applied magnetic field might easily affect which state, Q_{a+b} (q=1/3) or $\uparrow\uparrow\downarrow\downarrow$, CuFeO₂ adopts. It has been suggested that an O vacancy in a FeO₂ layer creates Fe²⁺ ions at the FeO₅ square pyramids

Table 1. Spin Exchange Energies (per FU) of the $\uparrow\uparrow\downarrow\downarrow$, Q_{a+b} (q=1/3,1/4,1/5), and $\uparrow\uparrow\uparrow\downarrow\downarrow$ states of CuFeO₂ and AgCrO₂ in Terms of Their Spin Exchange Parameters J_1-J_4

state	energy expression (per FU) ^a	CuFeO ₂ , meV/FU	AgCrO ₂ , meV/FU
↑ ↑↓↓	$(J_1 - J_2 + J_3 - 0.33J_4)(N^2/4)$	-1.58^{b}	-2.94^{c}
$Q_{a+b} (q = 1/3)$	$(1.50J_1 - 3J_2 + 1.50J_3)(N^2/4)$	-2.17	-7.01
$Q_{a+b} (q = 1/4)$	$(J_1 - J_2 + J_3 - 0.33J_4)(N^2/4)$	-1.58	-2.94
$Q_{a+b} (q = 1/5)$	$(0.19J_1 + 0.62J_2 + 1.31J_3 - 0.87J_4)(N^2/4)$	-0.77	+0.88
↑ ↑↑↓↓	$(0.20J_1 + 0.20J_2 + J_3 - 0.87J_4)(N^2/4)$	-0.49	+0.94

 $[^]aN = 5$ for CuFeO₂, and N = 3 for AgCrO₂. b Using the $J_1 - J_4$ values reduced by the overestimation factor f = 3.24 (see the text). c Using the $J_1 - J_4$ values reduced by the overestimation factor f = 1.42 (see the text).

around the vacancy site, and the uniaxial magnetism of such Fe²⁺ ions^{28,29} provides a driving force for the surrounding Fe³⁺ ions to orient their moments along the c-axis leading to the $\uparrow\uparrow\downarrow\downarrow$ state. An applied magnetic field tends to orient the moment of a magnetic ion perpendicular to the field when the field is not strong so that the Zeeman levels of the ion are almost equally populated. Thus, under such magnetic field applied along the c-axis, the spin orientation of CuFeO2 would be governed by two competing factors: the oxygen-vacancy effect favoring the //c-spin orientation and the magnetic-field effect favoring the ⊥c-spin orientation. Then, the adoption of the helical spiral-spin structure under magnetic field $H \approx 7-13$ T implies that the magnetic-field effect overcomes the oxygen-vacancy effect, so the intra- and interlayer spin frustration of CuFeO₂ is reduced by adopting the helical spiral-spin structure, as found for AgCrO₂. The occurrence of the $\uparrow\uparrow\uparrow\uparrow\downarrow\downarrow$ structure under H > \sim 13 T suggests that the energy gap between adjacent Zeeman levels of the Fe³⁺ ion become large, so only the lower-lying Zeeman levels are preferentially populated, thereby resulting in more moments parallel to the field direction.

5. CONCLUDING REMARKS

The spin exchanges of CuFeO₂ are frustrated, as found for AgCrO₂, but the two systems differ; the AFM interlayer exchange J_4 is strong in AgCrO₂ but weak in CuFeO₂, and the intralayer exchange J_2 is FM in AgCrO₂ but AFM in CuFeO₂. In contrast to the case of AgCrO₂, therefore, the $\uparrow\uparrow\downarrow\downarrow$ state is only slightly higher in energy than the helical spiral-spin state Q_{a+b} (q=1/3) in CuFeO₂ so that extrinsic factors such as oxygen vacancy and applied magnetic field can influence whether it adopts the $\uparrow\uparrow\downarrow\downarrow\downarrow$ or Q_{a+b} (q=1/3) structure.

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■ ACKNOWLEDGMENT

The research was supported by the Office of Basic Energy Sciences, Division of Materials Sciences, U.S. Department of Energy, under Grant No. DE-FG02- 86ER45259, and also by computing resources at the NERSC and the HPC Centers.

■ REFERENCES

(1) (a) Ederer, C.; Spaldin, N. A. Curr. Opin. Solid State Mater. Sci. 2005, 9, 128. (b) Ramesh, R.; Spaldin, N. A. Nat. Mater. 2007, 6, 21. (c) Khomskii, D. Physics 2009, 2, 20. (d) Dawber, M.; Rabe, K. M.; Scott, J. F. Rev. Mod. Phys. 2005, 77, 1083. (e) Eerenstein, W.; Mathur, N. D.; Scott, J. F. Nature (London) 2006, 442, 759. (f) Cheong, S. W.;

Mostovoy, M. Nat. Mater. 2007, 6, 13. (g) Tokura, Y. Science 2006, 312, 1481.

- (2) Kan, E.; Xiang, H.; Lee, C.; Wu, F.; Yang, J. L.; Whangbo, M. Angew. Chem., Int. Ed. 2010, 49, 1603.
- (3) (a) Mostovoy, M. Phys. Rev. Lett. 2006, 96, 067601. (b) Yamasaki, Y.; Miyasaka, S.; Kaneko, Y.; He, J.-P.; Arima, T.; Tokura, Y. Phys. Rev. Lett. 2006, 96, 207204. (c) Sergienko, I.; Sen, C.; Dagotto, E. Phys. Rev. Lett. 2006, 97, 227204. (d) Fiebig, M. J. Phys. D 2005, 38, R123. (e) Kimura, T.; Goto, T.; Shintani, H.; Ishizaka, K.; Arima, T.; Tokura, Y. Nature (London) 2003, 426, 55. (f) Hur, N.; Park, S.; Sharma, P. A.; Ahn, J. S.; Guha, S.; Cheong, S.-W. Nature (London) 2004, 429, 392.
- (4) (a) Chapon, L. C.; Radaelli, P. G.; Blake1, G. R.; Park, S.; Cheong, S.-W. *Phys. Rev. Lett.* **2006**, *96*, 097601. (b) Choi, Y. J.; Yi, H. T.; Lee, S.; Huang, Q.; Kiryukhin, V.; Cheong, S.-W. *Phys. Rev. Lett.* **2008**, *100*. 047601.
- (5) (a) Xiang, H. J.; Whangbo, M.-H. *Phys. Rev. Lett.* **2007**, 99, 257203. (b) Xiang, H. J.; Wei, S.-H.; Whangbo, M.-H.; Da Silva, J. L. F. *Phys. Rev. Lett.* **2008**, 101, 037209. (c) Zhang, Y.; Xiang, H. J.; M.-H. Whangbo, M.-H. *Phys. Rev. B* **2009**, 79, 054432. (d) Malashevich, A.; Vanderbilt, D. *Phys. Rev. Lett.* **2008**, 101, 037210. (e) Katsura, H.; Nagaosa, N.; Balatsky, A. V. *Phys. Rev. Lett.* **2005**, 95, 057205.
- (6) (a) Kimura, T.; Lashley, J. C.; Ramirez, A. P. Phys. Rev. B 2006, 73, 220401(R).
 (b) Nakajima, T.; Mitsuda, S.; Kanetsuki, S.; Prokes, K.; Podlensnyak, A.; Kimura, H.; Noda, Y. J. Phys. Soc. Jpn. 2007, 76, 043709.
 (c) Arima, T. J. Phys. Soc. Jpn. 2007, 76, 073702.
- (7) (a) Seki, S.; Onose, Y.; Tokura, Y. *Phys. Rev. Lett.* **2008**, *101*, 067204. (b) Kan, E. J.; Xiang, H. J.; Zhang, Y.; Lee, C.; Whangbo, M.-H. *Phys. Rev. B* **2009**, *80*, 104417.
- (8) (a) Prewitt, C. T.; Shannon, R. D.; Rogers, D. B. *Inorg. Chem.* **1971**, *10*, 719. (b) Pabst, A. *Am. Mineral.* **1946**, *31*, 539. (c) Soller, W.; Thompson, A. J. *Phys. Rev.* **1935**, *47*, 644.
- (9) Gehle, E.; Sabrowsky, H. Z. Naturforsch., B: Anorg. Chem. Org. Chem. 1975, 30, 659.
- (10) (a) Doumerc, J.-P.; Wichainchai, A.; Ammar, A.; Pouchard, M.; Hagenmuller, P. *Mater. Res. Bull.* **1986**, 21, 745. (b) El Ataoui, K.; Doumerc, J.-P.; Ammar, A.; Fournès, L.; Wattiaux, A.; Grenier, J.-C.; Pouchard, M. *J. Alloys Comp.* **2004**, 368, 79.
- (11) Mitsuda, S.; Mase, M.; Prokes, K.; Kitazawa, H.; Katori, H. A. J. Phys. Soc. Jpn. **2000**, 69, 3513.
- (12) (a) Mitsuda, S.; Mase, M.; Prokes, K.; Kitazawa, H.; Aruga Katori, H. *J. Phys. Soc. Jpn.* **2000**, *69*, 3513. (b) Petrenko, O. A.; Balakrishnan, G.; Lees, M. R.; Paul, D. McK.; Hoser, A. *Phys. Rev. B* **2000**, *62*, 8983.
- (13) (a) Kanetsuki, S.; Mitsuda, S.; Nakajima, T.; Anazawa, D.; Katori, H. A.; Prokes, K. *J. Phys.: Condens. Matter* **2007**, *19*, 145244. (b) Seki, S.; Yamasaki, Y.; Shiomi, Y.; Iguchi, S.; Onose, Y.; Tokura, Y. *Phys. Rev. B* **2007**, *75*, 100403(R). (c) Terada, N.; Nakajima, T.; Mitsuda, S.; Kitazawa, H.; Kaneko, K.; Metoki, N. *Phys. Rev. B* **2008**, *78*, 014101.
- (d) Mitsuda, S.; Nakajima, T.; Yamano, M.; Takahashi, K.; Yamazaki, H.; Masuda, K.; Kaneko, Y.; Terada, N.; Prokes, K.; Kiefer, K. *Phys. B* **2009**, 404, 2532.
- (14) Ye, F.; Fernandez-Baca, J. A.; Fishman, R. S.; Ren, Y.; Kang, H. J.; Qiu, Y.; Kimura, T. Phys. Rev. B 2007, 99, 157201.

(15) (a) Fishman, R. S. J. Appl. Phys. **2008**, 103, 07B109. (b) Fishman, R. S.; Ye, F.; Fernandez-Baca, J. A.; Haraldsen, J. T.; Kimura, T. Phys. Rev. B **2008**, 78, 140407(R). (c) Haraldsen, J. T.; Ye, F.; Fishman, R. S. Phys. Rev. B **2010**, 82, 144441.

- (16) (a) Terada1, N.; Narumi, Y.; Sawai, Y.; Katsumata, K.; Staub, U.; Tanaka, Y.; Kikkawa, A.; Fukui, T.; Kindo, K.; Yamamoto, T.; Kanmuri, R.; Hagiwara, M.; Toyokawa, H.; Ishikawa, T.; Kitamura1, H. Phys. Rev. B 2007, 75, 224411. (b) Lummen, T. T. A.; Strohm, C.; Rakoto, H.; Nugroho, A. A.; van Loosdrecht, P. H. M. Phys. Rev. B 2009, 80, 012406. (c) Lummen, T. T. A.; Strohm, C.; Rakoto, H.; van Loosdrecht, P. H. M. Phys. Rev. B 2010, 81, 224420. (d) Ye, Fe.; Ren, Y.; Huang, Q.; Fernandez-Baca, J. A.; Dai, P.; Lynn, J. W.; Kimura, T. Phys. Rev. B 2006, 73, 220404. (e) Eyert, V.; Frésard, R.; Maignan, A. Phys. Rev. B 2008, 78, 052402. (f) Malvestuto, M.; Bondino, F.; Magnano, E.; Lummen, T. T. A.; van Loosdrecht, P. H. M.; Parmigiani, F. Phys. Rev. B 2011, 83, 134422. (g) Plumer, M. L. Phys. Rev. B 2007, 76, 144411. (h) Plumer, M. L. Phys. Rev. B 2008, 78, 094402. (i) Quirion, G.; Tagore, M. J.; Plumer, M. L.; Petrenko, O. A. Phys. Rev. B 2008, 77, 094111.
- (17) (a) Blöchl, P. E. *Phys. Rev. B* **1994**, *50* (17), 953. (b) Kresse, G.; Joubert, D.**1999**, *59*, 1758.
- (18) Kresse, G.; Furthmüller J. Comput. Mater. Sci. 1996, 6, 15. Phys. Rev. B 1996, 54, 11169.
- (19) Perdew, J. P.; Burke, S.; Ernzerhof, M. Phys. Rev. Lett. 1996, 77, 3865.
- (20) (a) Dai, D.; Whangbo, M.-H. *J. Chem. Phys.* **2001**, *114*, 2887. (b) Dai, D.; Whangbo, M.-H. *J. Chem. Phys.* **2003**, *118*, 29. (c) Whangbo, M.-H.; Koo, H.-J.; Dai, D. *J. Solid State Chem.* **2003**, *176*, 417.
- (21) (a) Greedan, J. E. J. Mater. Chem. **2001**, 11, 37. (b) Dai, D.; Whangbo, M.-H. J. Chem. Phys. **2004**, 121, 672.
- (22) (a) Xiang, H. J.; Lee, C.; Whangbo, M.-H. *Phys. Rev. B: Rapid Commun.* **2007**, 76, 220411(R). (b) Koo, H.-J.; Whangbo, M.-H. *Inorg. Chem.* **2008**, 47, 128. (c) Koo, H.-J.; Whangbo, M.-H. *Inorg. Chem.* **2008**, 47, 4779.
- (23) Smart, J. S. Effective Field Theory of Magnetism; Saunders: Philadelphia, 1966.
- (24) Takeda, K.; Miyake, K.; Hitaka, M.; Kawae, T.; Ysguchi, N.; Mekata, M. J. Phys. Soc. Jpn. 1994, 63, 2017.
- (25) Olveira, G. N. P.; Lopes, A. M. L.; Mendonça, T. M.; Araújo, J. P.; Moreira, J. A.; Almeida, A.; Amaral, V. S.; Correia, J. G. *Hyperfine Interact.* **2010**, *197*, 123.
- (26) Oohara, Y.; Mitsuda, S.; Yoshizawa, H.; Yaguchi, N.; Kuriyama, H.; Asana, T.; Mekata, M. *J. Phys. Soc. Jpn.* **1994**, *63*, 847.
- (27) Seki, S.; Onose, Y.; Tokura, Y. Phys. Rev. Lett. 2008, 101, 067204.
- (28) Whangbo, M.-H.; Dai, D.; Lee, K.-S.; Kremer, R. K. Chem. Mater. 2006, 18, 1266.
 - (29) Dai, D.; Whangbo, M.-H. Inorg. Chem. 2005, 44, 4407.