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## Communications

### Cyanonaphthalene Diimide Semiconductors for Air-Stable, Flexible, and Optically Transparent n-Channel Field-Effect Transistors

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Organic semiconductors offer potentially inexpensive active components in large-area and flexible optoelectronics such as complementary circuits (CMOS), light-emitting diodes (OLEDs), and photovoltaics (OPVs).<sup>1</sup> Because efficient charge transport in organic semiconductors is thought to proceed via hopping involving delocalized  $\pi$ -orbitals, such materials tend to be highly conjugated and thus excellent chromophores with optical extinction coefficients ( $\epsilon$ ) of  $\sim 10^4$  to  $\sim 10^6$  M<sup>-1</sup> cm<sup>-1</sup>. However, materials for transparent displays or charge-blocking layers in OLEDs/OPVs should ideally be transparent to visible light, requiring a  $>3$  eV band gap.<sup>2</sup> A common strategy for achieving wide band gap chromophores is to compress the molecular conjugation

length;<sup>3</sup> however, this frequently depresses charge-carrier mobility.<sup>4</sup> Consequently, attempts to fabricate transparent organic field-effect transistors (OFETs) with high mobility have generally yielded low band gap ( $<2.5$  eV) films with significant visible absorption.<sup>5</sup>

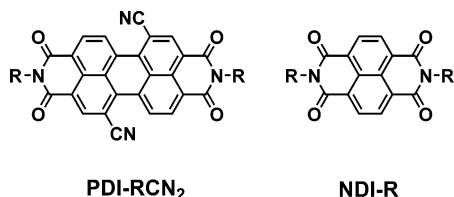
Recent studies of dicyanoperylene diimides (PDI-RCN<sub>2</sub>)<sup>6</sup> have demonstrated a unique combination of high electron mobility (as high as 0.6 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>), environmental stability, and solution processability.<sup>6b</sup> Thus, PDI-RCN<sub>2</sub>s yield complementary organic logic and frequency-generating devices with unprecedented performance.<sup>6c-e</sup> However, the  $\sim 2.4$  eV PDI-RCN<sub>2</sub> band gap and  $\epsilon = 47000$  M<sup>-1</sup> cm<sup>-1</sup> renders 50 nm films intensely red to the eye and unsuitable for transparent organic optoelectronics.<sup>6a,6b</sup>

Previous research on core-unsubstituted naphthalene diimide (NDI) semiconductors demonstrated that this wide band gap ( $\sim 3$  eV) materials class can also exhibit high electron mobility, environmental stability, and solution

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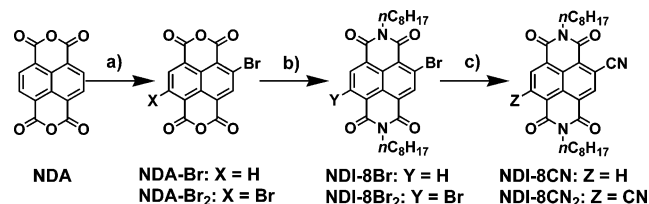
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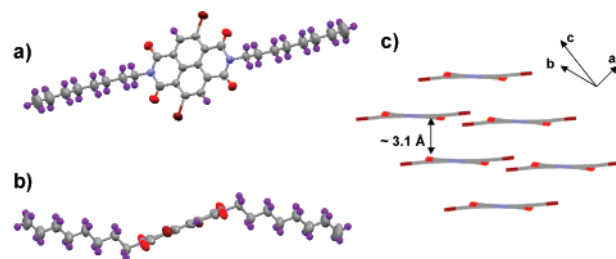
processability, but not in a single material.<sup>7</sup> NDI air stability typically requires the presence of fluorinated *N*-R groups, which also results in depressed mobility ( $0.06 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ ) relative to *N*-*n*-octyl NDI ( $0.16 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ ).<sup>7b</sup> In this communication, we report two new core-cyanated naphthalene diimide semiconductors, NDI-8CN and NDI-8CN<sub>2</sub> (Scheme 1),

**Scheme 1. Synthetic Route to NDI-8CN and NDI-8CN<sub>2</sub>, Where the Reaction Conditions Are: (a) Br<sub>2</sub>/I<sub>2</sub>, Oleum; (b) *n*-octyl Amine, HOAc; (c) CuCN, DMF.**

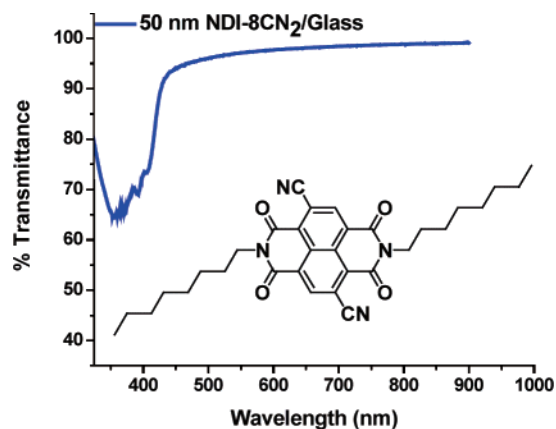


which represent the first air-stable, high-mobility, and transparent organic *n*-type semiconductors. Electrical properties are evaluated in bottom-gate (Si/SiO<sub>2</sub>) top-contact (Au) OFETs, along with thin film microstructure and morphology. Finally, the first visible region transparent OFET channel is fabricated. The syntheses of NDI-8CN and NDI-8CN<sub>2</sub> are achieved via a new NDI core bromination, cyanation sequence.

Typically, core-substituted NDIs are accessed via pyrene chlorination;<sup>8</sup> however, in the present work, NDA is brominated with I<sub>2</sub>/Br<sub>2</sub> to yield a mixture of monobrominated (NDA-Br) and dibrominated (NDA-Br<sub>2</sub>) products. Condensation with *n*-octyl amine is accomplished by refluxing in acetic acid, and the resulting mono- and dibromoisimides can be readily separated chromatographically. Interestingly, only a single substitutional isomer of NDI-8Br<sub>2</sub> is isolated, as evidenced by <sup>1</sup>H NMR and single-crystal X-ray diffraction (Figure 1), whereas perylene dianhydride dibromination yields a mixture of substitutional isomers.<sup>8d</sup> Previous rylene imide cyanations relied on Pd catalysts with Zn(CN)<sub>2</sub>,<sup>6a,6b</sup> however, cyanation of NDI-8Br and NDI-8Br<sub>2</sub> is accomplished with CuCN in DMF to afford NDI-8CN and NDI-8CN<sub>2</sub> in ~45% cyanation yield, without air-sensitive Pd catalysts.



**Figure 1.** NDI-8Br<sub>2</sub> crystal structure depicting (a) the face-on view demonstrating the one substitutional isomer, (b) the side view depicting a nearly planar naphthalene core, and (c) the packing diagram demonstrating the small interplanar intermolecular distance of ~3.1 Å. *N,N'*-groups have been removed for clarity.



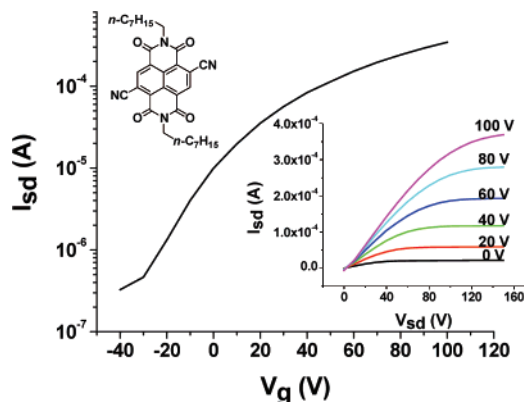
**Figure 2.** Transmission optical spectrum of a 50 nm vapor-deposited thin film of NDI-8CN<sub>2</sub> on glass demonstrating the impressive transparency of this material between 400 and 800 nm.

NDI-8CN and NDI-8CN<sub>2</sub> electronic structures were examined by cyclic voltammetry, optical spectroscopy, and photoluminescence. Electrochemical reduction potentials in dichloromethane vs S.C.E. are  $-0.22 \text{ V}$  for NDI-8CN and  $+0.08 \text{ V}$  for NDI-8CN<sub>2</sub>, consistent with systematic LUMO energy depression with increasing cyanation. Importantly, NDI-8CN<sub>2</sub> has a reduction potential similar to that of PDI-RCN<sub>2</sub> ( $-0.07 \text{ V}$  vs S.C.E.);<sup>6a-b</sup> therefore, the LUMO/charge-carrier energies in the NDI and PDI materials should be similar. Optical and photoluminescence spectroscopy of these NDI derivatives reveals a band gap of ~3 eV, reflecting the smaller conjugated core dimensions relative to PDIs.<sup>6a,6b</sup> Thus, thin films of these NDIs are transparent in the visible region (Figure 2).

Thin (50 nm) NDI-8CN and NDI-8CN<sub>2</sub> films were grown by physical vapor deposition ( $2 \times 10^{-6}$  Torr,  $0.2 \text{ Å/s}$ ) onto doped Si substrates having a 300 nm thermally grown SiO<sub>2</sub> dielectric. During film deposition, the growth temperature (*T<sub>d</sub>*) was varied to optimize the semiconductor film microstructure/morphology. All thin films were characterized by OFET measurements, X-ray diffraction (XRD), and tapping-mode AFM. Top-contact OFETs with  $100 \text{ μm}/5 \text{ mm}$  *S/D* width/length were fabricated by thermally depositing 50 nm thick gold electrodes onto the NDI films through a shadow mask.

OFET measurements performed in vacuum ( $\sim 10^{-6}$  Torr) reveal optimal average electron mobilities for NDI-8CN and NDI-8CN<sub>2</sub> films of  $4.7 \times 10^{-3} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  and  $0.15 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ , for *T<sub>d</sub>* values of 130 and 110 °C, respectively. These differences in mobility despite similar chemical structures

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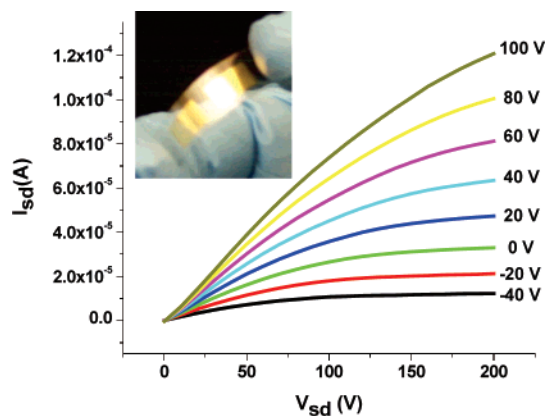


**Figure 3.**  $I$ – $V$  curves measured in air for NDI-8CN<sub>2</sub> films ( $T_d = 130$  °C) on Si/SiO<sub>2</sub> substrates after storage in ambient atmosphere for 5 months. The transfer plot yields an electron mobility of  $0.1 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  and  $I_{\text{on}}/I_{\text{off}} = 10^3$ . The inset of the output plot demonstrates the well-defined linear and saturation regime at the gate bias ( $V_g$ ) indicated above the trace.

are discussed with the AFM and XRD data below. Interestingly, OFET operation in ambient atmosphere reveals that the NDI-8CN devices undergo severe degradation of  $I$ – $V$  characteristics, whereas the NDI-8CN<sub>2</sub> devices exhibit stable operation with only a slightly lower maximum average mobility of  $0.11 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  (Figure 3). The NDI-8CN<sub>2</sub> robustness suggests that ambient stability can be extended to other rylene diimides via polycyanation to achieve reduction potentials  $\sim 0 \text{ V}$  vs S.C.E. The current on–off ratios ( $I_{\text{on}}/I_{\text{off}}$ ) can be as high as  $\sim 10^5$  for NDI-8CN and  $\sim 10^3$  for NDI-8CN<sub>2</sub> thin films. The lower  $I_{\text{on}}/I_{\text{off}}$  ratio of NDI-8CN<sub>2</sub> is due to high  $I_{\text{off}}$  ( $\sim 1 \times 10^{-6} \text{ A}$ ), which is likely due to dopants in the NDI-8CN<sub>2</sub> thin films or proximate dielectric layer.

AFM reveals similar polycrystalline morphologies for NDI-8CN and NDI-8CN<sub>2</sub> films, with ribbon-like grains until  $T_d \approx 90$  °C, and plate-like grains at higher  $T_d$  settings. XRD measurements on NDI-8CN and NDI-8CN<sub>2</sub> films indicate similar, highly textured microstructures, exhibiting only 00 $l$  reflections and with a  $d$ -spacing of  $18.2 \text{ \AA}$ . The primary difference in the film XRD data for the two materials is the presence of a second family of Bragg reflections in NDI-8CN<sub>2</sub> films grown at  $T_d > 90$  °C, corresponding to a  $d$ -spacing of  $20.2 \text{ \AA}$ . However, there is no obvious correlation between mobility and these additional reflections. Given the similar morphologies and microstructures of both NDI-8CN and NDI-8CN<sub>2</sub> films, the difference in mobility in vacuum is likely related to in-plane ordering, which cannot be rigorously evaluated with the present  $\theta/2\theta$  XRD and AFM data.

Top-contact bottom-gate transparent channel flexible n-type OFETs were fabricated with NDI-8CN<sub>2</sub> to demonstrate the unique materials properties. Thin NDI-8CN<sub>2</sub> films (50



**Figure 4.** Output plot for a transparent, flexible OFET having a PEDOT:PSS gate, polymer gate dielectric, NDI-8CN<sub>2</sub> semiconductor, and Au source and drain, and exhibiting an electron mobility of  $0.03 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  in air. Inset: photograph of an array of  $\sim 100$  devices fabricated on overhead transparency film demonstrating transparency and flexibility.

nm) were vapor-deposited onto overhead transparency film coated with a spin-cast PEDOT:PSS polymeric gate and a P-UV-013 polymer dielectric.<sup>9</sup> Next, 20 nm gold source/drain electrodes were evaporated through a shadow mask onto the NDI-8CN<sub>2</sub> films to give an OFET of  $S/D$  width/length =  $100 \mu\text{m}/5 \text{ mm}$ . This air-stable, flexible, transparent OFET exhibits a mobility of  $0.03 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ ,  $V_{\text{th}} = -2 \text{ V}$ , and  $I_{\text{on}}/I_{\text{off}} \sim 10^3$  in ambient atmosphere (Figure 4). An analogous rigid device fabricated on an ultrasMOOTH ITO/glass substrate as a gate gives  $\mu = 0.08 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ ,  $V_{\text{th}} = 4 \text{ V}$ , and  $I_{\text{on}}/I_{\text{off}} \sim 10^3$  in ambient atmosphere.

In summary, new NDI core halogenation chemistry yields the first cyano NDIs. Dicyanation affords a similar reduction potential to PDI-RCN<sub>2</sub>s and imparts air-stability to the fluorine-free semiconductor. Utilizing PEDOT:PSS or ITO as the gate electrode and a polymer dielectric, the first high-mobility, air-stable, n-type transparent channel OFETs were fabricated with NDI-8CN<sub>2</sub>.

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**Supporting Information Available:** Experimental details; spectral, electrochemical, film diffraction, AFM, electrical data, OFET fabrication details; CIF file. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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(9) Details of transparent device fabrication are contained in the Supporting Information.