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## Shape-Dependent Reducibility of Cuprous Oxide Nanocrystals

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In this paper the reducibility of octahedral and cubic Cu<sub>2</sub>O nanocrystals that respectively expose the (111) and (100) crystal planes has been investigated both experimentally and theoretically. Reduced either by H<sub>2</sub> or CO, the reduction temperature of octahedral Cu<sub>2</sub>O nanocrystals is lower than that of cubic Cu<sub>2</sub>O nanocrystals by more than 120 °C, which provides the unambiguous experimental results that the Cu<sub>2</sub>O(111) crystal plane is much more facile to be reduced than the Cu<sub>2</sub>O(100) crystal plane. The DFT calculation results reveal that the different reducibilities of octahedral and cubic Cu<sub>2</sub>O nanocrystals arise from their different surface structures, in which the Cu<sub>2</sub>O(111) surface has coordination unsaturated Cu but the Cu<sub>2</sub>O(100) surface does not. The chemisorption and activation of CO and H<sub>2</sub> are stronger on the Cu<sub>2</sub>O(111) surface than on the Cu<sub>2</sub>O(100) surface. These results provide the convincing evidence for the shape-dependent surface reactivity of oxide nanocrystals.

## 1. Introduction

Engineering the surface of heterogeneous catalysts is critical to the design of new processes that minimize the number and amount of unwanted byproducts and that operate with high energy efficiency. Therefore, great efforts have been aimed at fundamentally understanding the surface structure–catalytic performance relationships and tuning the catalytic performance by controlling the surface structure of catalyst particles. The surface structure of a particle relies on the crystal planes that it exposes. In the past few decades, extensive surface science studies on bulk single crystals of metals and oxides under the ultrahigh vacuum conditions have identified that the variation in reactivity often depends on the exposed crystal plane.<sup>1</sup> However, technical catalysts are usually supported nanosized metal particles or metal oxide particles with poorly defined surface structure, which hampers the understanding of the surface structure–catalytic performance relationships and also may lead to the nonideal catalytic performance. The recent great advances in nanoscience enable the controllable synthesis of uniform nanocrystals with a well-defined shape that preferentially expose one type of crystal plane,<sup>2</sup> which can nicely serve as model systems for the fundamental investigations of the surface structure–catalytic performance relationship at the nanoscale and under “working” conditions and are also likely highly active and selective catalysts.<sup>3</sup> The shape-dependent catalytic performance has been reported for Pt,<sup>4</sup> Rh,<sup>5</sup> Ru,<sup>6</sup> Fe,<sup>7</sup> Ag<sup>8</sup> metal nanocrystals and CeO<sub>2</sub>,<sup>9</sup> Co<sub>3</sub>O<sub>4</sub><sup>10</sup> oxide nanocrystals.

Cuprous oxide (Cu<sub>2</sub>O) has been used as a catalyst for several important oxidation reactions.<sup>11</sup> Uniform cubic and octahedral Cu<sub>2</sub>O nanocrystals that respectively expose the (100) and (111) crystal plane have been successfully synthesized.<sup>12</sup> It has been

reported that the octahedral Cu<sub>2</sub>O nanocrystals show a higher activity in the photodegradation of rhodamine B and methylene blue molecules than the cubic Cu<sub>2</sub>O nanocrystals.<sup>13</sup> In this paper, we report convincing experimental results of the shape-dependent reducibility of Cu<sub>2</sub>O nanocrystals. Reduced either by H<sub>2</sub> or CO, octahedral Cu<sub>2</sub>O nanocrystals are much more facile to be reduced than cubic Cu<sub>2</sub>O nanocrystals. The accompanying DFT calculation results demonstrate that the different reducibilities of octahedral and cubic Cu<sub>2</sub>O nanocrystals arise from their different surface structures. Since the reducibility of an oxide is closely related to its catalytic performance in oxidation reactions, our results infer that the catalytic activity of Cu<sub>2</sub>O nanocrystals in oxidation reactions may depend on their shape and that octahedral Cu<sub>2</sub>O nanocrystals are likely more efficient catalysts.

## 2. Experimental Section

The synthesis of Cu<sub>2</sub>O nanocrystals followed Zhang et al.’s recipe.<sup>12f</sup> Typically, 5.0 mL of NaOH aqueous solution (2.0 mol·L<sup>-1</sup>) was added dropwise into 50 mL of CuCl<sub>2</sub> aqueous solution (0.01 mol·L<sup>-1</sup>) containing given amounts of poly(vinylpyrrolidone) (PVP, *M*<sub>w</sub> = 30 000) at 55 °C. After adequately stirring for 0.5 h, 5.0 mL of ascorbic acid aqueous solution (0.6 mol·L<sup>-1</sup>) was added dropwise into the solution. The mixed solution was adequately stirred for 3 h at 55 °C. The resulting precipitate was collected by centrifugation and decanting, followed by washing with distilled water and absolute ethanol, and finally dried in vacuum at RT for 12 h.

Powder X-ray diffraction (XRD) patterns were recorded on a Philips X’Pert PROS diffractometer using a nickel-filtered Cu Kα (wavelength: 0.154 18 nm) radiation source with the operation voltage and operation current being 40 kV and 50 mA, respectively. X-ray photoelectron spectroscopy (XPS) measurements were performed on an ESCALAB 250 high-performance electron spectrometer using monochromatized Al Kα (*hν* = 1486.7 eV) as the excitation source. The likely charging of samples was corrected by setting the binding energy

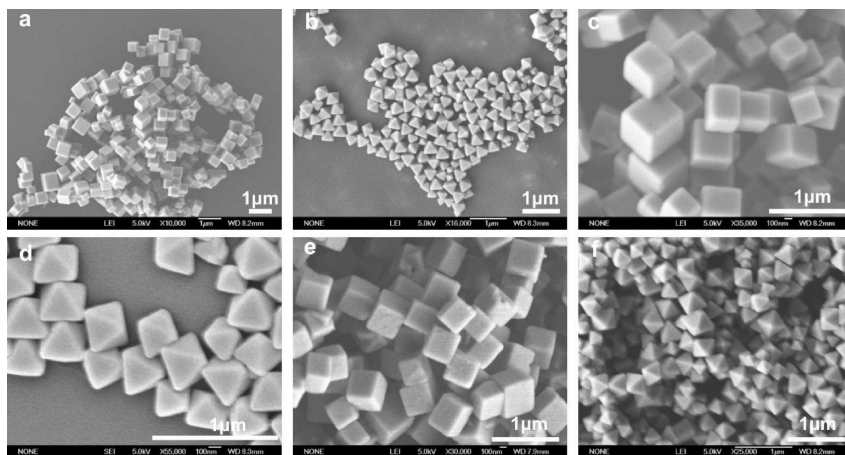
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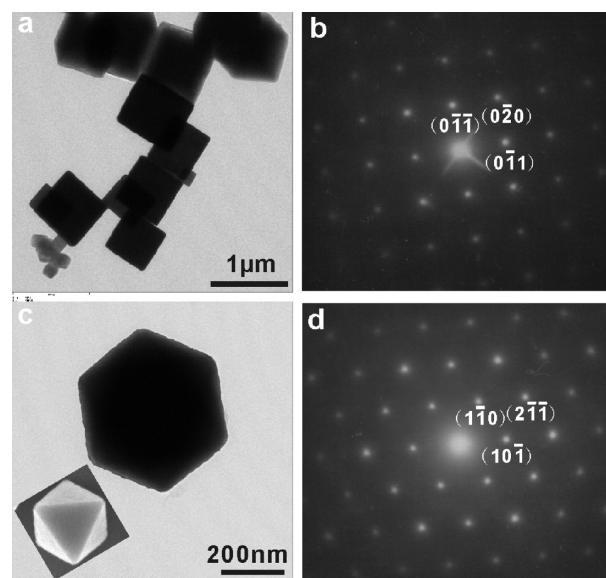


**Figure 1.** Representative SEM images of as-synthesized cubic (a and c) and octahedral (b and d) Cu<sub>2</sub>O nanocrystals, cubic Cu<sub>2</sub>O nanocrystals calcined in N<sub>2</sub> at 300 °C for 1 h (e), and octahedral Cu<sub>2</sub>O nanocrystals calcined in N<sub>2</sub> at 200 °C for 1 h (f). The scale bar in all SEM images is 1 μm.

of the adventitious carbon (C 1s) to 284.5 eV. Scanning electron microscope (SEM) experiments were performed on a JEOL JSM-6700 field emission scanning electron microscope. Transmission electron microscopy (TEM) and selected area electron diffraction (ED) experiments were performed on a JEOL-2010 high-resolution transmission electron microscope. The reducibility of Cu<sub>2</sub>O nanocrystals was investigated by the temperature-programmed reduction (TPR) technique employing H<sub>2</sub> (H<sub>2</sub>-TPR) or CO (CO-TPR) as the reducing gas. Typically, 10 mg of catalyst was placed in a quartz reactor and heated at a rate of 5 °C/min. In H<sub>2</sub>-TPR, 5% H<sub>2</sub> balanced with Ar with a flow rate of 20 mL/min was used and the consumption of H<sub>2</sub> was measured by a thermal conductivity detector (TCD). In CO-TPR, 1% CO balanced with Ar with a flow rate of 40 mL/min was used and the CO<sub>2</sub> product was online analyzed by a HIDEN QIC-20 gas analysis system. The reactivity of the adventitious carbon and likely organic compounds on the surface of Cu<sub>2</sub>O nanocrystals was also studied by means of temperature-programmed reaction spectroscopy (TPRS) in which 10 mg of Cu<sub>2</sub>O nanocrystals was heated in a quartz reactor in Ar (flow rate of 30 mL/min) at a heating rate of 5 °C/min and all likely products (CO<sub>2</sub>, H<sub>2</sub>O, H<sub>2</sub>, CO, N<sub>2</sub>, NH<sub>3</sub>, NO, and NO<sub>2</sub>) were online analyzed by a HIDEN QIC-20 gas analysis system.

### 3. Theoretical Calculation Methods

Theoretical calculations were performed within the spin-polarized density functional theory implemented in the DMol package.<sup>14</sup> The double-numeric quality basis set with polarization functions (DNP) was used. DFT semicore pseudopotential (DSPP) and PBE functional<sup>15</sup> were used for all the atoms. A thermal smearing of 0.002 hartree and a real-space cutoff of 4.4 Å were adopted. In our calculations, the unpolarized Cu<sub>2</sub>O(111) was represented by four O–Cu–O layers with oxygen-terminated surfaces, and the polarized Cu<sub>2</sub>O(100) surface was represented by four O–Cu–O–Cu layers with one oxygen-terminated surface and another Cu-terminated surface in order to keep the stoichiometry ratio. The bottom two O–Cu–O or O–Cu–O–Cu layers were fixed at their bulk coordination. The vacuum space along the *z* direction was set as 18 Å, which was long enough to screen the interactions from image atoms. Our test calculations indicate that the *k*-points mesh with space of 0.06 Å<sup>−1</sup> was dense enough to give the converged energy. The adsorption of CO and H<sub>2</sub> was calculated on the relaxed oxygen-terminated surfaces.

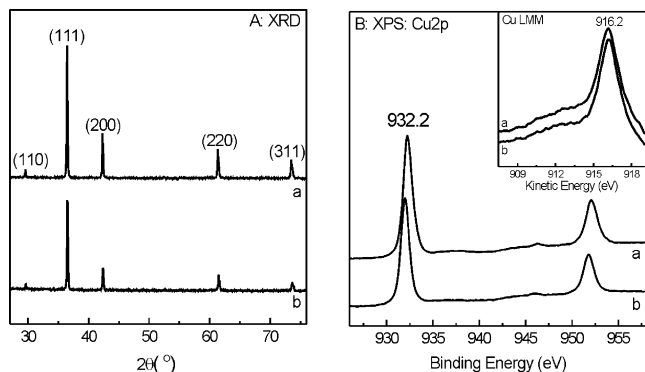


**Figure 2.** TEM images and corresponding electron diffraction patterns of as-synthesized cubic (a and b) and octahedral (c and d) Cu<sub>2</sub>O nanocrystals. The inset shows the SEM image of one octahedral Cu<sub>2</sub>O nanocrystal that corresponds to the TEM image.

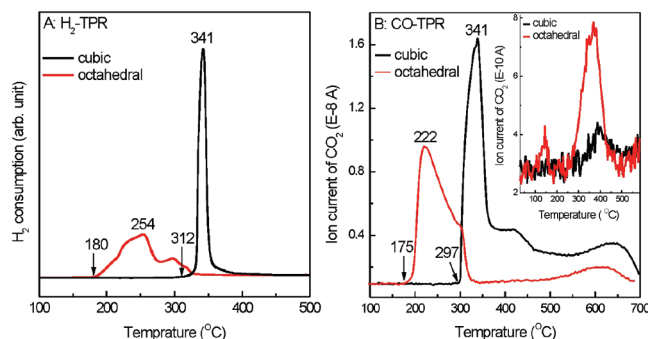
### 4. Results and Discussion

Following Zhang et al.'s recipe,<sup>12f</sup> Cu<sub>2</sub>O nanocrystals with a uniform cubic shape (Figure 1a and c) and those with a uniform octahedral shape (Figure 1b and d) were successfully synthesized with the molar ratio of PVP to CuCl<sub>2</sub> being 0 and 80, respectively. The cubic Cu<sub>2</sub>O nanocrystals mostly have a size of 400–700 nm and the octahedral Cu<sub>2</sub>O nanocrystals mostly have a size of ~300 nm. The TEM images and corresponding ED patterns (Figure 2) clearly reveal that cubic and octahedral Cu<sub>2</sub>O nanocrystals expose the (100) and (111) surfaces, respectively, agreeing with previous reports.<sup>12</sup> The powder XRD patterns of both octahedral and cubic nanocrystals (Figure 3A) can be well indexed to Cu<sub>2</sub>O (JCPDS file number 78-2076). The XPS results (Figure 3B) show that the Cu 2p<sub>3/2</sub> binding energy and the Cu LMM Auger peak kinetic energy of both samples are located at 932.2 and 916.2 eV, respectively, demonstrating that their surfaces remain Cu<sub>2</sub>O.<sup>16</sup>

Interestingly, the reducibilities of octahedral and cubic Cu<sub>2</sub>O nanocrystals differ significantly. Figure 4A shows the H<sub>2</sub>-TPR



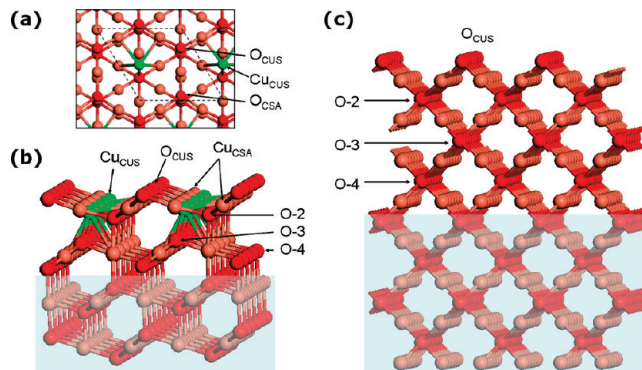
**Figure 3.** XRD (A) and Cu 2p XPS spectra (B) of as-synthesized cubic (a) and octahedral (b)  $\text{Cu}_2\text{O}$  nanocrystals. The inset shows the corresponding Cu LMM Auger peaks.



**Figure 4.**  $\text{H}_2$ -TPR (A) and CO-TPR (B) spectra of as-synthesized  $\text{Cu}_2\text{O}$  octahedral and cubic nanocrystals. The inset shows the  $\text{CO}_2$ -TPRS spectra of octahedral and cubic  $\text{Cu}_2\text{O}$  nanocrystals in Ar.

spectra in which the  $\text{H}_2$  consumption peak corresponds to the reduction of  $\text{Cu}_2\text{O}$  by  $\text{H}_2$  to produce  $\text{H}_2\text{O}$ . The cubic  $\text{Cu}_2\text{O}$  nanocrystals give a single reduction peak beginning at  $\sim 312^\circ\text{C}$  and reaching the maximum rate at  $341^\circ\text{C}$ ; however, the octahedral  $\text{Cu}_2\text{O}$  nanocrystals give multiple reduction peaks beginning at  $\sim 180^\circ\text{C}$  and reaching the maximum rate at  $254^\circ\text{C}$ . The onset reduction temperature of octahedral  $\text{Cu}_2\text{O}$  nanocrystals is lower by  $132^\circ\text{C}$  than that of cubic  $\text{Cu}_2\text{O}$  nanocrystals. The CO-TPR spectra (Figure 4B) in which the  $\text{CO}_2$  peak corresponds to the reduction of  $\text{Cu}_2\text{O}$  by CO show the similar tendency. The reduction of cubic  $\text{Cu}_2\text{O}$  nanocrystals begins at  $\sim 297^\circ\text{C}$  and reaches the maximum rate at  $341^\circ\text{C}$ , but the reduction of octahedral  $\text{Cu}_2\text{O}$  nanocrystals begins at  $\sim 175^\circ\text{C}$  and reaches the maximum rate at  $222^\circ\text{C}$ , and the reduction temperature of octahedral  $\text{Cu}_2\text{O}$  nanocrystals is lower by  $122^\circ\text{C}$  than that of cubic  $\text{Cu}_2\text{O}$  nanocrystals. The TPR results clearly demonstrate that  $\text{Cu}_2\text{O}$  octahedral nanocrystals are much more facile to be reduced either by  $\text{H}_2$  or CO than  $\text{Cu}_2\text{O}$  cubic nanocrystals. The peak areas in the  $\text{H}_2$ -TPR and CO-TPR are similar for both octahedral and cubic  $\text{Cu}_2\text{O}$  nanocrystals, and their reduction reaction in CO and  $\text{H}_2$  begins at the similar temperatures; however, the reduction reaction of  $\text{Cu}_2\text{O}$  in  $\text{H}_2$  finishes before  $400^\circ\text{C}$ , whereas that in CO extends to  $700^\circ\text{C}$ . This indicates that there might be some differences between the reduction mechanism of  $\text{Cu}_2\text{O}$  by  $\text{H}_2$  and that by CO, which needs further investigations.

The existence of adventitious carbon on the surface of particles is usually inevitable for the synthesis of powder samples. The XPS results show that the surface atomic concentration of adventitious carbon is 35% for cubic  $\text{Cu}_2\text{O}$  nanocrystals. Due to the use of PVP in the synthesis recipe, besides adventitious carbon, PVP was also detected on the

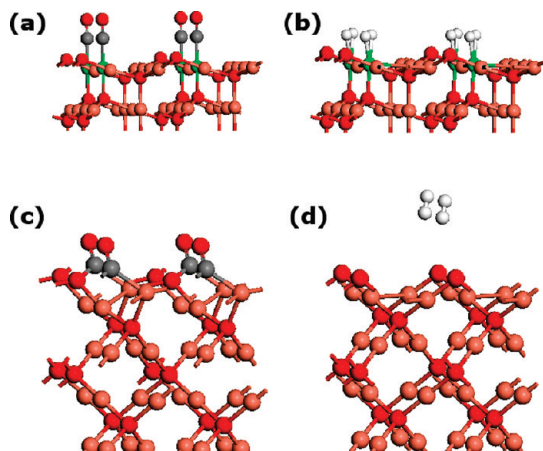


**Figure 5.** Optimized structures of  $\text{Cu}_2\text{O}(111)$  and  $(100)$  surfaces: (a) top, (b) side view of  $\text{Cu}_2\text{O}(111)$  surface, and (c) side view of  $\text{Cu}_2\text{O}(100)$  surface. The shadow parts show the atoms fixed during the geometric optimization. The red, brick red, and green balls represent oxygen, coordinated saturated copper ( $\text{Cu}_{\text{CSA}}$ ), and coordinated unsaturated copper ( $\text{Cu}_{\text{CUS}}$ ) atoms, respectively.

surface of octahedral  $\text{Cu}_2\text{O}$  nanocrystals. The XPS results show that the surface atomic concentrations of carbon and nitrogen are 39% and 3.13% for octahedral  $\text{Cu}_2\text{O}$  nanocrystals, respectively. To check the influence of these surface carbon-containing species, we studied the reactivity of adventitious carbon and likely organic compounds on the surface of  $\text{Cu}_2\text{O}$  nanocrystals by means of TPRS in Ar. Only the formation of  $\text{CO}_2$  was obviously observed (the inset in Figure 4B), and more  $\text{CO}_2$  was produced from octahedral  $\text{Cu}_2\text{O}$  nanocrystals than from cubic  $\text{Cu}_2\text{O}$  nanocrystals, agreeing with the XPS results that the surface atomic concentration of adventitious carbon on octahedral  $\text{Cu}_2\text{O}$  nanocrystals is higher than that on cubic  $\text{Cu}_2\text{O}$  nanocrystals. However, the  $\text{CO}_2$  peaks in the  $\text{CO}_2$ -TPRS spectra are weaker in intensity by 2 orders of magnitude than those in the CO-TPR spectra, implying that the adventitious carbon and likely organic compounds do not likely contribute to the significant difference between the reducibility of octahedral and cubic  $\text{Cu}_2\text{O}$  nanocrystals. We also found that the shapes and sizes of octahedral and cubic  $\text{Cu}_2\text{O}$  nanocrystals (Figure 1e and f) can be well preserved after calcination at high temperatures at 200 and  $300^\circ\text{C}$  in  $\text{N}_2$  for 1 h, respectively. This is reasonable because the melting point of  $\text{Cu}_2\text{O}$  crystals is  $1232.0^\circ\text{C}$ . These observations demonstrate that the surface structures of octahedral and cubic  $\text{Cu}_2\text{O}$  nanocrystals do not likely change prior to the occurrence of reduction reaction. Therefore, the great difference between the reducibility of octahedral and cubic  $\text{Cu}_2\text{O}$  nanocrystals should arise from their different surface structures, i.e., different shapes.

The reduction of an oxide always initiates from the surface. The octahedral and cubic  $\text{Cu}_2\text{O}$  nanocrystals expose the (111) and (100) crystal planes, respectively. Therefore, our results demonstrate that the  $\text{Cu}_2\text{O}(111)$  surface is much more facile to be reduced than the  $\text{Cu}_2\text{O}(100)$  surface. DFT calculations were performed to understand the structures and reactivity of  $\text{Cu}_2\text{O}(111)$  and  $\text{Cu}_2\text{O}(100)$  surfaces. Figure 5 illustrates the optimized  $\text{Cu}_2\text{O}(111)$  and  $\text{Cu}_2\text{O}(100)$  surface structures. Both stable surfaces are O-terminated. On the  $\text{Cu}_2\text{O}(111)$  surface, four types of atoms exist: coordinated unsaturated Cu ( $\text{Cu}_{\text{CUS}}$ ) and O ( $\text{O}_{\text{CUS}}$ ) and coordinated saturated Cu ( $\text{Cu}_{\text{CSA}}$ ) and O ( $\text{O}_{\text{CSA}}$ ). The distance between the nearest-neighboring  $\text{Cu}_{\text{CUS}}$  and  $\text{O}_{\text{CSA}}$  ( $d(\text{Cu}_{\text{CUS}}-\text{O}_{\text{CSA}})$ ), the nearest-neighboring  $\text{Cu}_{\text{CUS}}$  and  $\text{Cu}_{\text{CSA}}$  ( $d(\text{Cu}_{\text{CUS}}-\text{Cu}_{\text{CSA}})$ ), and the nearest-neighboring  $\text{Cu}_{\text{CSA}}$  and  $\text{O}_{\text{CUS}}$  ( $d(\text{Cu}_{\text{CSA}}-\text{O}_{\text{CUS}})$ ) is 1.91, 2.51, and  $1.83 \text{ \AA}$ , respectively; and the distance along the  $z$  direction between the nearest-neighboring  $\text{Cu}_{\text{CSA}}$  and  $\text{Cu}_{\text{CUS}}$  ( $z(\text{Cu}_{\text{CSA}}-\text{Cu}_{\text{CUS}})$ ) is  $0.23 \text{ \AA}$ . On





**Figure 6.** Most stable adsorption configurations for CO and H<sub>2</sub> adsorption on the Cu<sub>2</sub>O(111) and (100) surfaces: (a) CO on the Cu<sub>2</sub>O(111) surface; (b) H<sub>2</sub> on the Cu<sub>2</sub>O(111) surface; (c) CO on the Cu<sub>2</sub>O(100) surface; (d) H<sub>2</sub> on the Cu<sub>2</sub>O(100) surface. The red, brick red, green, gray, and white balls represent oxygen, Cu<sub>CSA</sub>, Cu<sub>CUS</sub>, carbon, and hydrogen atoms, respectively.

the Cu<sub>2</sub>O(100) surface, however, only three types of atoms exist: O<sub>CUS</sub>, O<sub>CSA</sub>, and Cu<sub>CSA</sub>. The distance between the nearest-neighboring Cu<sub>CSA</sub> and O<sub>CUS</sub> ( $d(\text{Cu}_{\text{CSA}}-\text{O}_{\text{CUS}})$ ) is 1.76 Å.

The chemisorptions of CO and H<sub>2</sub> on the Cu<sub>2</sub>O(111) and (100) surfaces were calculated, and Figure 6 illustrates the most stable configurations. On the Cu<sub>2</sub>O(111) surface, CO prefers to chemisorb on the Cu<sub>CUS</sub> top site while it adsorbs on the Cu<sub>CSA</sub>—O<sub>CUS</sub> bridge site of the Cu<sub>2</sub>O(100) surface. The adsorption energy of CO (1.52 eV) on the Cu<sub>2</sub>O(111) surface is larger than that (1.12 eV) on the Cu<sub>2</sub>O(100) surface. H<sub>2</sub> chemisorbs on the Cu<sub>CUS</sub> top site of the Cu<sub>2</sub>O(111) surface but only physisorbs on the Cu<sub>2</sub>O(100) surface. The adsorption energy of H<sub>2</sub> (0.41 eV) on the Cu<sub>2</sub>O(111) surface is much larger than that (0.03 eV) on the Cu<sub>2</sub>O(100) surface, and the H—H bond length of H<sub>2</sub>(ads) is 0.80 Å on the Cu<sub>2</sub>O(111) surface but 0.74 Å on the Cu<sub>2</sub>O(100) surface, suggesting that H<sub>2</sub> is highly activated upon the chemisorption on the Cu<sub>2</sub>O(111) surface. These results suggest that the Cu<sub>2</sub>O(111) surface is generally more active toward the chemisorption of reactants than the Cu<sub>2</sub>O(100) surface, which could be reasonably attributed to the existence of Cu<sub>CUS</sub> on the Cu<sub>2</sub>O(111) surface. The chemisorption of CO and H<sub>2</sub> on the oxide surfaces is the first step in the reduction of oxide.<sup>17</sup> Therefore, although detailed DFT calculations are needed for a thorough understanding of the reduction behaviors of Cu<sub>2</sub>O nanocrystals, it can be adequately concluded that the different surface structures of octahedral and cubic Cu<sub>2</sub>O nanocrystals result in their different reducibilities.

The shape-dependent reducibility of Cu<sub>2</sub>O nanocrystals is among the most unambiguous evidence for the shape-dependent reactivity of transitional metal oxides; moreover, its origin is successfully clarified at the molecular level. Zhou et al.<sup>9a</sup> reported that CeO<sub>2</sub> nanorods preferentially exposing (001) and (110) crystal planes are more active in CO oxidation than the irregular CeO<sub>2</sub> nanoparticles. Si and Flytzani-Stephanopoulos<sup>9b</sup> reported that the Au/CeO<sub>2</sub> catalyst with CeO<sub>2</sub> nanorods is much more active than that with CeO<sub>2</sub> nanocubes in the water—gas shift reaction. The catalytic activity of Co<sub>3</sub>O<sub>4</sub> nanocrystals with different shapes was reported to follow the following order: nanosheets predominantly exposing the (112) crystal plane > nanobelts predominantly exposing the (011) crystal plane >> nanocubes predominantly exposing the (001) crystal plane.<sup>10a</sup> Xie et al.<sup>10b</sup> recently reported the excellent catalytic performance

of Co<sub>3</sub>O<sub>4</sub> nanorods predominantly exposing the (110) crystal plane with rich Co(III) in low-temperature CO oxidation. All these results vindicate the novel concept that the reactivity and catalytic activity of transitional metal oxides can be controlled and tuned by controlling their shape, which opens up a strategy for both the improvement of heterogeneous catalysts without the change of the catalyst composition and the design of highly efficient heterogeneous catalyst. We believe that the catalytic activity of Cu<sub>2</sub>O nanocrystals in oxidation reactions is shape-dependent and that the octahedral Cu<sub>2</sub>O nanocrystals are likely efficient catalysts, which are under investigation.

## 5. Conclusion

In summary, the octahedral Cu<sub>2</sub>O nanocrystals exposing the (111) crystal plane are much more facile to be reduced either by H<sub>2</sub> or CO than the cubic Cu<sub>2</sub>O nanocrystals exposing the (100) crystal plane. The different reducibilities of octahedral and cubic Cu<sub>2</sub>O nanocrystals arise from their different surface structures, in which the Cu<sub>2</sub>O(111) surface has coordination unsaturated Cu but the Cu<sub>2</sub>O(100) surface does not. Therefore, the Cu<sub>2</sub>O(111) surface exhibits a higher activity toward the chemisorption and activation of CO and H<sub>2</sub> than the Cu<sub>2</sub>O(100) surface. These results provide the convincing evidence for the shape-dependent surface reactivity of oxide nanocrystals.

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