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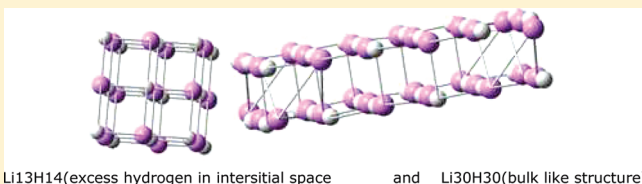
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Structural Evolution and Stability of Hydrogenated Li_n ($n = 1-30$) Clusters: A Density Functional Study

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ABSTRACT: The structure and electronic and optical properties of hydrogenated lithium clusters Li_nH_m ($n = 1-30$, $m \leq n$) have been investigated by density functional theory (DFT). The structural optimizations are performed with the Becke 3 Lee–Yang–Parr (B3LYP) exchange–correlation functional with 6-311G++(d, p) basis set. The reliability of the method employed has been established by excellent agreement with computational and experimental data, wherever available. The turn over from two- to three-dimensional geometry in Li_nH_m clusters is found to occur at size $n = 4$ and $m = 3$. Interestingly, a rock-salt-like face-centered cubic structure is seen in $\text{Li}_{13}\text{H}_{14}$. The sequential addition of hydrogen to small-sized Li clusters predicted regions of regular lattice in saturated hydrogenated clusters. This led us to focus on large-sized saturated clusters rather than to increase the number of hydrogen atoms monotonically. The lattice constants of Li_9H_9 , $\text{Li}_{18}\text{H}_{18}$, $\text{Li}_{20}\text{H}_{20}$, and $\text{Li}_{30}\text{H}_{30}$ calculated at their optimized geometry are found to gradually approach the corresponding bulk values of 4.083. The sequential addition of hydrogen stabilizes the cluster, irrespective of the cluster size. A significant increase in stability is seen in the case of completely hydrogenated clusters, i.e., when the number of hydrogen atoms equals Li atoms. The enhanced stability has been interpreted in terms of various electronic and optical properties like adiabatic and vertical ionization potential, HOMO–LUMO gap, and polarizability.



INTRODUCTION

Lithium clusters are the most studied clusters in theory, and hydrogenated lithium clusters are important as models due to their low number of electrons. Li_nH_m clusters appear to be the ideal prototype to study interaction of hydrogen atoms with a metal and chemisorption of hydrogen on the surface of a metal. The nature of bonding involved has been the subject of intense theoretical and experimental research. Experimental measurements of binding energy, ionization potential, and dissociation channels of hydrogenated lithium clusters are available. Theoretical calculations have provided valuable information about the geometry and other properties of the mentioned clusters. The results are explained by the hypothesis that the bonding of one hydrogen atom localizes one of the valence electrons of the Li cluster, the other electrons remaining delocalized. It has been observed that the binding of hydrogen to clusters is favored compared to that in corresponding crystals because of availability of interstitial space, lack of lattice strain energy, and lower electron density in the cluster.^{1,2} The present work is based on calculations of the equilibrium structures of lithium clusters with varying hydrogen content and cluster size. Recent reports that the lithium-based metal system is a promising material for efficient hydrogen storage^{3–5} have given an additional boost to their study with an aim to ascertain if LiH clusters can be efficient hydrogen storage devices. Such devices would play a crucial role in designing technologically important hydrogen storing media.

In the past two decades, mass spectroscopy and other techniques have made available experimental results on LiH clusters.

Antoine et al.² have studied the dissociation pathway and binding energies of the Li_nH_m^+ clusters by using a reflectron mass spectrometer. These clusters are found to decompose into LiH and Li_2H_2 molecules by evaporation. The metal–insulator segregation in lithium-rich Li_nH_m^+ clusters has also been investigated by Antoine et al.⁶ Vezin et al.⁷ recorded the first vibrational and rotational spectra of Li_2H and optical absorption spectra of Li_4H and Li_9H .⁸ It has been confirmed that the hydrogen atom assumes a peripheral position bridging two and three Li atoms in the planar and three-dimensional structure of Li_4H and Li_9H , respectively. The photoionization measurements of ionization potential of Li_nH_m clusters by Vezin et al.^{9,10} suggest that Li clusters tend to lose metallic character as the number of hydrogen atom increases.

A variety of quantum mechanical calculations have been performed to investigate the Li_nH_m cluster. Kato et al.¹¹ used the ab initio theory at the Hartree–Fock (HF) level to study the structure and stability of small Li_nH_m ($m \leq n \leq 4$). Jena et al.^{12,13} have employed the self-consistent molecular orbital method in the HF approximation to study structural and electronic properties of pure and hydrogenated Li clusters. They found that although the electronic character of Li and H in LiH is of ionic nature the electronic structure of LiH is not completely ionic. Jena et al.^{14,15} also studied structural and electronic properties of

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Li_nH_m ($m \leq n \leq 3$) clusters and reported the difference between cluster–hydrogen and metal–hydrogen interactions. The coupled cluster computations by Shaefer et al.^{16,17} show excellent agreement with available experimental data⁷ for the atomization energies and ionization potentials of small Li_n and Li_nH ($n = 1-4$) clusters. Geometry optimizations at multireference configuration interaction (MRCI) level using large basis sets on neutral and cationic Li_nH_m ($n = 1-7$, $m = 1,5$) clusters have been performed,^{18,19} and high-level ab initio molecular orbital theory has also been employed²⁰ to calculate the geometries, vibrational frequencies, atomic charges, and binding energies of small clusters (LiH)_{*n*}, (NaH)_{*n*}, (BeH_2)_{*n*}, and (MgH_2)_{*n*}.

Since it is impractical to employ robust ab initio electronic structure methods to larger-sized clusters, recent studies on hydrogenated Li clusters are primarily based on density functional theory (DFT) which is computationally more accessible. Fuentealba et al. have conducted DFT^{21,22} based studies of Li_nH_m ($n \leq 4$, $m \leq 4$) clusters with the focus on static dipole polarizability and electron localization function to analyze the bonding characteristic. Bertolus et al.²³ presented a study of Li_nH_n and $\text{Li}_{n+1}\text{H}_n^+$ (n up to 7) and $\text{Li}_{14}\text{H}_{13}^+$ using both DFT and Model Potential, and their results have shown that Li–H is almost totally ionic. Wu and Jones²⁴ have presented a joint experimental/DFT study of the enthalpies of reaction for the hydrogenated Li clusters. The study of sequential hydrogen binding energies on model Li_{18} clusters ($\text{Li}_{18}\text{H}_{2n+2}$, $n = 0-8$) has also been done by employing the DFT method.²⁵

Up to now, though there are several measurements and theoretical researches examining the properties of Li_nH_m clusters, there remain some open questions: How does hydrogenation change the properties of Li clusters? What are the preferred occupation sites of hydrogen in Li_nH_m clusters? Whether a transition toward solid state can be observed in the cluster size studied here? The current work presents a comprehensive study of sequential binding of hydrogen on a Li cluster by varying hydrogen content and cluster size. Larger clusters containing dozens of Li and hydrogen atoms are probably investigated here for the first time to observe the transition to solid state.

COMPUTATIONAL METHOD

The computations have been carried out within the DFT framework. The significant advantage of the DFT method lies in incorporating electronic correlation at a much lower cost than standard chemistry methods through an effective exchange correlation potential. The exchange correlation functional, B3LYP, is a widely used hybrid functional comprised of Becke's three-parameter exchange functional²⁶ and the Lee–Yang–Parr correlation functional.²⁷ It has been shown that B3LYP yields good results on structures and energetics of ion-covalent clusters²⁸ and has been successfully employed to investigate the ground state geometries and electronic structure calculations of the Li_nH_m cluster.^{15,21,23,25} Hadipour et al.²⁹ have employed the B3LYP functional with 6-311G++(2d, 2p) basis set to predict geometries, binding energies, atomic charges,⁷ Li nuclear magnetic shielding, and EFG tensors of small Li_nH_m ($m \leq n \leq 4$) clusters. The present work employs the B3LYP functional with 6-311G++(d, p) basis set; the double plus version in the basis set adds the diffuse function to the hydrogen and Li atom; and the p function is added for the H atom and d function for the Li atom. The calculations have been carried out using the Gaussian 03 program package.³⁰ It is pleasing to find that our DFT

calculations reliably predict the structure evolution of large-sized Li_nH_m clusters toward the bulk LiH crystal.

Numerous papers^{15,18,19,21,23} have been devoted to finding global minima of small Li_nH_m clusters. The present work does not attempt to find the lowest energy structure among the different isomers of Li_nH_m clusters. The optimized geometry of pure Li_n clusters was taken as a starting structure to build the hydrogenated lithium cluster, Li_nH_m . Hydrogen atoms were added one by one to the optimized Li_n cluster geometry until it became completely saturated. Literature was referred to while generating starting geometries for small-sized Li_nH_m ($1 \leq n \leq 4$, $1 \leq m \leq 4$) clusters. Subsequent inputs were generated based on the optimized geometries of $\text{Li}_n\text{H}_{m-1}/\text{Li}_{n-1}\text{H}_m$. Frequency analysis was also performed over the optimized geometry at the same level of theory to check for no imaginary frequency. For the cluster Li_nH_m of size ($m, n \leq 9$), the default self-consistent field convergence of 10^{-8} is used, and for a larger cluster size ($m > 9, n > 9$) the 10^{-6} convergence limit is employed. The ground state of each of the Li_nH_m clusters has the lowest possible multiplicity that can be obtained for a given number of unpaired electrons. All electronic and optical calculations have been performed over optimized geometries.

RESULTS AND DISCUSSION

I. Structural Evolution of the Li_nH_m Cluster toward the Face-Centered Cubic Bulk LiH Crystal. The geometries of small neutral hydrogenated lithium clusters Li_nH_m ($n \leq 9, m \leq 9$) are shown in Figure 1(a). For LiH, the calculated bond length in the current work, i.e., 1.594 Å, is close to the experimental bond length (1.595 Å).³¹ The previously theoretically reported values are 1.64,^{15,21} 1.61,¹⁸ and 1.595 Å¹⁶ calculated by DFT,^{15,21} SCF,¹⁸ and CCSD¹⁶ methods, respectively. Ionization extends the bond length of LiH to 2.19 Å because of Coulomb repulsion between Li and H atoms having 0.5 e charge on each. The optimized geometry of Li_2H is an obtuse isosceles triangle, and the Li–Li and Li–H distances are 2.55 and 1.70 Å, respectively. These bond lengths are in agreement with those obtained by multiphoton ionization spectroscopy (2.52 and 1.71 Å)⁷ and the previous calculations.^{16,18,21} The geometry of the lowest energy structure of Li_2H_2 is a rhombus with Li–Li and Li–H bond lengths as 2.25 and 1.74 Å, respectively. These results are also in good agreement with the reported values.^{19,21} It establishes that the level of theory employed in the current work strikes a fair balance between computational effort and accuracy. The initial structure of the Li_3H cluster was formed by replacing one Li atom in the Li_4 cluster (a 60° rhombus) with a H atom, which resulted in a distorted rhombus structure after optimization, accompanied by a shortening of Li–Li bond length in the Li–H–Li subunit. A similar structure has been reported previously.^{16,18,21} The optimized geometry of Li_3H_2 can be viewed as a deformed trapezoidal. The optimized geometry of Li_3H_3 is an equilateral triangle with Li atoms at the vertices and H atoms at the center of the sides, Li–Li and Li–H bond lengths being 2.75 and 1.71 Å, respectively. A similar structure has been reported in the literature.^{18,25,20} The optimized geometry of Li_4H_4 has T_d symmetry with Li–H bond length as 1.83 Å.^{19–22,29} The Li_4H_4 cluster having eight valence electrons (sufficient to fill completely the set of degenerate one electron level) is a closed shell system. The highly symmetrical T_d structure of this cluster is possible in the absence of Jahn–Teller distortion.³² Analysis of the vibrational frequencies by Chen et al.²⁰ shows that with

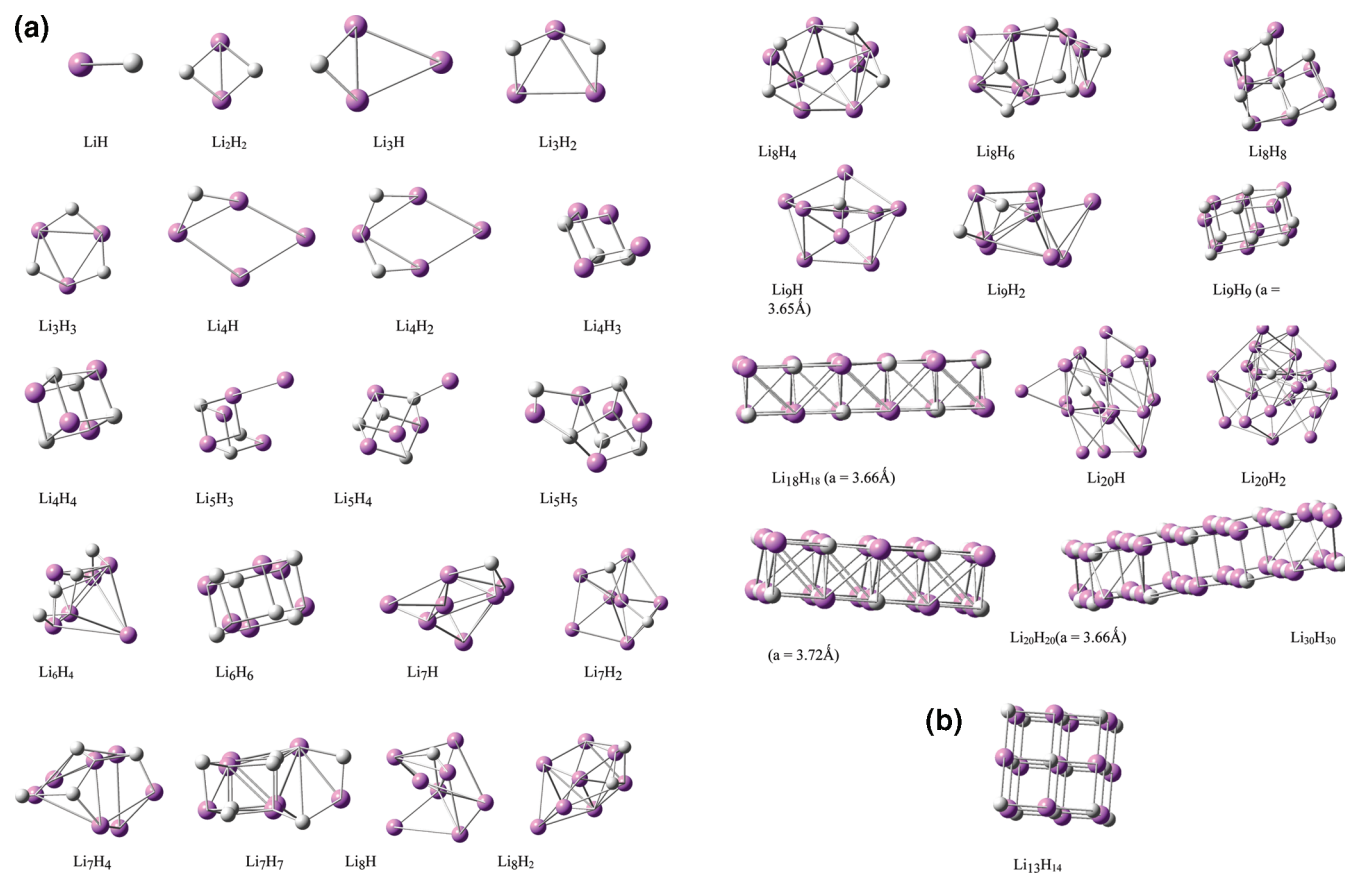


Figure 1. (a) Optimized geometries of neutral hydrogenated lithium Li_nH_m cluster ($1 \leq n, m \leq 30$) and of Li_9H_9 , $\text{Li}_{18}\text{H}_{18}$, $\text{Li}_{20}\text{H}_{20}$, and $\text{Li}_{30}\text{H}_{30}$ clusters with their lattice constants. (b) Optimized geometry of $\text{Li}_{13}\text{H}_{14}$.

T_d symmetry Li_4H_4 has tetracentred bonding and is more rigid when compared to the tricentred bonded structure with D_{4h} symmetry. The optimized geometry of Li_5H_3 is a distorted cube with F-center vacancy and also has a tail of a Li atom attached to it; the same has been reported by Koutecky et al.¹⁹ The optimized geometry of Li_5H_4 is a distorted cube with a tail of a Li atom; a similar structure is found for its cationic counterpart, in contradiction to the structure reported by Bertolus et al.²³ For Li_5H_5 we find a chair-type structure, and Li_6H_4 has a caplike structure, in agreement with Koutecky et al.¹⁹

The optimized Li_6H_6 cluster has the shape of a hexagonal unit cell. The input for the geometry optimization of Li_7H was generated by capping a hydrogen atom on the pentagonal bipyramid structure of Li_7 ; the distorted pentagon bipyramid with hydrogen over one of its triangular faces is obtained as the optimized structure of Li_7H . The optimized geometry of Li_8H_8 has a chair-type structure. In Li_9H , the H atom occupies an interstitial position between two Li atoms. For the Li_9H_9 cluster, nine H atoms were introduced as a cap over the triangular faces of the twisted bcc Li_9 cluster. The optimized structure of the Li_9H_9 cluster looks like an fcc unit cell with the H atoms occupying the interstitial positions between Li atoms. $\text{Li}_{13}\text{H}_{14}$ is stable with a NaCl-like fcc structure (see Figure 1b) with binding energy per atom as 2.12 eV. It seems that availability of interstitial space has made room for excess hydrogen. Most atoms in the cluster are surface atoms, so hydrogen is readily trapped. Lui et al.³³ have performed a theoretical search within the DFT framework and concluded that the interstitial position is the preferential position of the H atom in the bulk Li

which has bcc structure. The position of the hydrogen atom in Li_nH_m clusters studied here is in accordance with the conclusion drawn by Lui et al.³³ It indicates that the behavior of LiH clusters tends to approach the bulk behavior at small size.

This encouraged us to generate the fcc coordinates of $\text{Li}_{18}\text{H}_{18}$, $\text{Li}_{20}\text{H}_{20}$, and $\text{Li}_{30}\text{H}_{30}$ for geometry optimization. It was worthwhile to check whether the lattice constants (distance between the nearest like ions) of these cluster approach toward that of bulk lattice. Interestingly, the optimized geometries of Li_9H_9 , $\text{Li}_{18}\text{H}_{18}$, $\text{Li}_{20}\text{H}_{20}$, and $\text{Li}_{30}\text{H}_{30}$ have the FCC structure (see Figure 1a) with an averaged value of lattice constants as 3.65, 3.66, 3.66, and 3.72 Å, respectively. The lattice constant of the bulk crystal structure of LiH is 4.084 Å.⁶ It concludes that the structure of Li_nH_m clusters in the nano range is the same as that of the bulk crystal structure, albeit with a relatively smaller lattice constant.

II. Properties of Li_nH_m Clusters. A. *Binding Energy, Energy of HOMO Level, and HOMO–LUMO Gap.* Table I displays the values of binding energy per atom of Li_nH_m cluster.

$$E_{\text{BE}}/(n+m) = [nE(\text{Li}) + mE(\text{H}) - E(\text{Li}_n\text{H}_m)]/(n+m)$$

Addition of hydrogen to the Li_n cluster increases the binding energy, and a significant increase is observed when the number of H atoms is equal to the number of Li atoms, i.e., Li_nH_n cluster. This is because of the fact that hydrogen localizes the valence electrons of the Li atom, and the saturated Li_nH_n has a closed shell. There is charge transfer from the Li 2s to hydrogen 1s state. As a result, the energy of the HOMO level of the cluster gets lowered. This is reflected in the electronic behavior of the cluster

Table I. Binding Energy Per Atom ($E_{BE}/(m + m)$), Vertical and Adiabatic Ionization Potentials (VIP/AIP), Energy of HOMO Level (E_H), HOMO–LUMO Gap (E_g), and Static Dipole Polarizability (α) of the Li_mH_m Cluster at B3LYP/6-311G++(d,p)^a

cluster size	$E_{BE}/(m + m)$	VIP/AIP (EXP ^{e,f})	$-E_H$	E_g	α
LiH	1.26 (1.207 ^b , 1.10 ^d)	8.04/7.99(7.70 ^e)			28.06
Li ₂ H _m					
$m = 1$	1.20 (1.15 ^b , 1.38 ^c , 1.06 ^d)	4.79/4.16(4.10 ^e)	3.03	1.53	200.98
$m = 2$	1.78 (2.03 ^c , 1.61 ^d)	8.26/7.21	5.71	5.03	33.41
Li ₃ H _m					
$m = 1$	1.22 (1.22 ^b , 1.33 ^c , 1.13 ^d)	5.31/5.02(4.67 ^e , 4.74 ^f)	3.58	2.23	179.61
$m = 2$	1.59 (1.79 ^c , 1.45 ^d)	5.34/4.09(4.62 ^e , 4.45 ^f)	3.38	2.46	129.94
$m = 3$	1.95 (2.17 ^c , 1.78 ^d)	8.68	6.39	5.78	49.37
Li ₄ H _m					
$m = 1$	1.12 (1.13 ^b , 1.22 ^c)	4.69/4.14(4.10 ^e , 4.17 ^f)	3.23	1.87	296.32
$m = 2$	1.54 (1.54 ^c)	3.62/4.68(5.00 ^{e,f})	3.81	2.56	179.20
$m = 3$	1.79 (1.95 ^c , 1.63 ^d)	/4.12(4.38 ^{e,f})	2.94	2.05	135.55
$m = 4$	2.04 (2.22 ^c , 1.86 ^d)	8.45/7.99	6.30	5.68	65.13
Li ₅ H _m					
$m = 3$	1.73 (1.58 ^d)	5.27/4.93(4.90 ^{e,f})	3.59	2.62	202.28
$m = 4$	1.86 (1.70 ^d)	4.28, 4.21(4.33 ^e)	2.86	1.74	244.94
$m = 5$	2.05	8.07/6.76	6.01	5.24	79.8
Li ₆ H _m					
$m = 4$	1.80 (1.65 ^d)	5.27/4.66	3.67	2.48	230.84
$m = 6$	2.11	8.16/7.12(4.50 ^f)	6.29	5.47	91.58
Li ₇ H _m					
$m = 1$	1.12	4.35/4.20(4.20 ^{e,f})	3.12	1.52	482.36
$m = 2$	1.33	4.30/3.74(4.10 ^{e,f})	3.01	1.52	462.90
$m = 4$	1.70	4.25/3.64(4.18 ^f)	2.86	1.47	314.85
$m = 7$	2.09	7.92/6.09	5.96	5.07	107.99
Li ₈ H _m					
$m = 1$	1.07	3.76/3.51(3.97 ^e , 3.93 ^f)	2.62	1.29	620.55
$m = 2$	1.32	4.26/4.04(4.17 ^f)	3.08	1.51	752.15
$m = 4$	1.66	4.55/3.98(4.35 ^f)	3.23	1.86	365.37
$m = 6$	1.94	5.20	3.60	2.53	242.85
$m = 8$	2.10	7.49/6.03	5.77	4.83	121.82
Li ₉ H _m					
$m = 1$	1.08	3.91/3.81(4.19 ^e , 4.18 ^f)	2.78	1.20	621.50
$m = 2$	1.29	3.95/3.73(4.01 ^f)	2.77	1.34	
$m = 9$	2.16	7.55/7.01	5.89	5.10	293.04
Li ₁₈ H ₁₈					
	2.22		3.84	2.87	
Li ₂₀ H _m					
$m = 1$	1.10		2.61	1.03	
$m = 2$	1.18		2.83	1.09	
$m = 20$	2.12		5.39	4.43	
Li ₃₀ H ₃₀					
	2.25		4.40	4.40	

^a Except, α (in au) all values are in eV. ^b Ref 16. ^c Ref 21. ^d Ref 19. ^e Ref 10. ^f Ref 9.

like reactivity and conductance. The energy of the HOMO level (E_H) is also listed in Table I. For example, see the value of E_H for

Li₄H_n at $n = 1, 2, 3$, and 4 in Table I; E_H of Li₄H₄ has significantly lowered, making it less reactive.

In accordance with the trend observed in binding energy, the HOMO–LUMO gap (E_g) increases as H atoms are added to the Li_n cluster and particularly at the saturated level, i.e., Li_nH_m shows a sudden jump. Li_nH_m clusters with an even number of electrons have a higher band gap:²⁴ E_g for Li_4H , Li_4H_2 , and Li_4H_3 is 1.87, 2.56, and 2.50 eV, respectively. The Li_mH_n clusters with an even number of electrons have a closed shell and are hence more stable than the ones with an odd number of electrons.²⁴ However, this effect is not pronounced in larger sized clusters as the E_g for Li_{20}H_2 (1.09) and Li_{20}H (1.03) does not differ significantly.

B. Vibrational Frequency of Li_nH_m Clusters. Frequency analysis was performed over the optimized geometry at the same level of DFT theory with 6-311G++(d, p). The vibrational frequencies thus computed followed the same trend for Li_nH_m ($n = 1-4$, $m = 1-4$) as that reported in the work of Chen et al.²⁰ However, for higher sizes ($n = 5-9$, $m = 5-9$), there is no regular trend at the higher frequency part, as for $(\text{LiH})_5$, $(\text{LiH})_6$, $(\text{LiH})_7$, $(\text{LiH})_8$, and $(\text{LiH})_9$ higher frequency modes are 1224, 1153, 1238, 1323, and 1272, respectively (see Table II). This is because as cluster size increases all the Li–H bond lengths are not the same and hence no general comment over the trend of vibrational frequencies with increase of cluster size can be made.

C. Static Dipole Polarizability. Static dipole polarizability (α) is a measure of extent of distortion of electron density, and therefore α is very sensitive to the delocalization of electrons. It is an important observable for understanding the electronic properties of clusters such as E_g and Ionization Potential (IP). According to simple perturbation theory, α should be inversely correlated to the E_g ³⁴ (see Table I). Values of α in the case of Li_4H_m ($m = 1-4$) are decreasing, whereas their corresponding E_g values are increasing with hydrogenation. This is valid for all sizes of the Li_nH_m ($n = 1-9$, $m = 1-9$) cluster.

Polarizability can also be related to the cluster geometry. The increase in coordination number with cluster size leads to a decrease in polarizability.³⁵ As in the case of Li_4H_m , geometry becomes more compact as m increases from 1 to 4. Li_4H and Li_4H_2 have planar geometry, whereas Li_4H_3 and Li_4H_4 have cubic geometry (see Figure 1a). Their α value falls from 296.32 (at $m = 1$) to 65.13 (at $m = 4$), and it is true for all Li_n clusters as more hydrogen atoms are added until the saturation (see Table I). Hence, we may conclude that polarizability decreases as clusters become more compact with increasing size. The above conclusion can be validated by arguing that since the polarizability per atom is a quantity related to the volume per atom of cluster the more compact structures have smaller volume and in turn have relatively few and short bonds. This leads to tighter bonding of their valence electrons. Hence their electron density is not distorted easily which means low α value. Thus, successive addition of hydrogen to Li_n clusters lowers their polarizability.

The results are analyzed as a function of cluster size and composition. In the case of saturated clusters, Li_nH_m ($n = m$), volume increases as the cluster size, i.e., n increases (see Figure 1(a)), and as a result α increases as shown in Figure 2. The saturated hydrogenated lithium clusters Li_nH_n with minimum polarizability are more stable than the metal-rich ($m < n$) cluster. This is in accordance with the minimum polarizability principal (MPP) that any system evolves toward a state of minimum polarizability.^{36–38}

D. Adiabatic and Vertical Ionization Potential. The computed values of ionization potential (IP) are listed in Table I and compared with the experimental data^{9,10} whenever possible. The vertical ionization potential (VIP) measures the energy difference between the ground state of the neutral and ionized cluster with frozen geometry of the neutral clusters, whereas adiabatic

Table II. Frequencies (cm^{-1}) of the Li_nH_m Cluster at 6-311G++(d, p)

LiH	(LiH) ₂	(LiH) ₃	(LiH) ₄	(LiH) ₅	(LiH) ₆	(LiH) ₇	(LiH) ₈	(LiH) ₉
1406	1177	1234	1097	1224	1153	1238	1323	1272
	1080	1231	966	1208	1150	1181	1254	1243
	971	1214	965	1075	1139	1167	1239	1219
	888	1003	962	995	1134	1148	1178	1173
	601	944	944	962	1125	1119	1105	1091
	519	940	942	933	1009	1113	1082	1081
		592	937	917	1006	1042	1053	1064
		413	917	819	900	996	1008	1000
		326	911	789	895	929	988	990
		325	707	732	872	924	959	979
		262	704	731	855	902	934	964
		258	700	664	821	866	903	915
			520	491	815	847	894	910
			391	426	784	827	840	903
			390	408	718	795	809	893
			388	381	711	783	783	854
			370	373	681	730	764	817
			370	364	673	714	718	805
				261	438	707	667	770
				233	423	668	635	767
				209	416	451	631	752
				153	405	447	615	741
					368	417	538	688
					362	411	516	629
					326	378	479	613
					305	347	449	558
					242	335	441	482
					239	326	340	477
					182	292	410	461
					173	237	375	454
						232	340	427
						224	325	424
						209	311	386
						163	291	374
						148	277	363
						114	260	356
							231	344
							214	339
							173	294
							156	288
							145	287
							43	283
								259
								249
								246
								245
								231
								189

ionization potential (AIP) is a measure of the difference between the energies of the appropriate neutral form and the cationic one at their optimized geometries. It can be seen that the experimental values lie well within the predicted values of AIP and VIP.

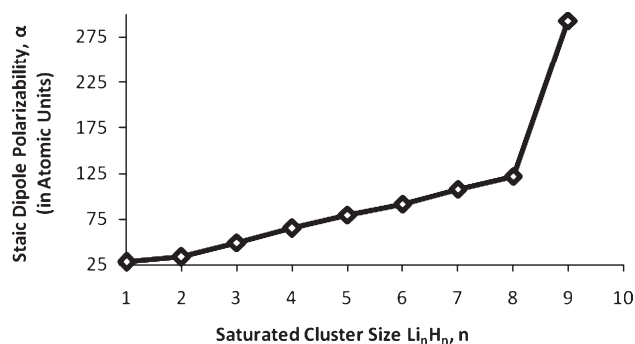


Figure 2. Static dipole polarizability (α) vs cluster size of the neutral saturated hydrogenated Li_nH_m cluster ($1 \leq n \leq 9$, $1 \leq m \leq 9$).

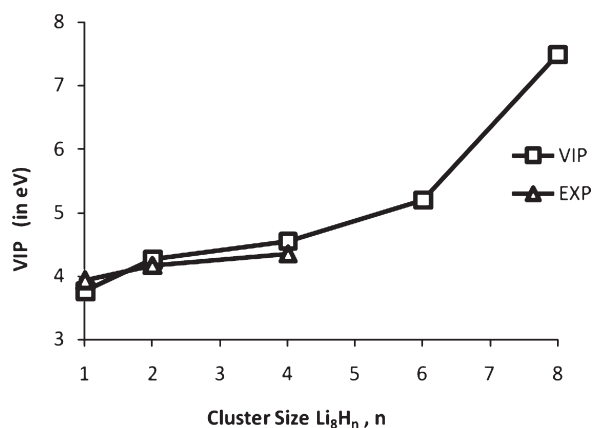


Figure 3. Vertical ionization potential (AIP and VIP) vs cluster size of the neutral saturated hydrogenated Li_nH_m cluster ($1 \leq n \leq 9$, $1 \leq m \leq 9$).

VIP of Li_nH_n clusters increases with hydrogenation (see Figure 3). Hydrogenation of lithium clusters (Li_n) induces the hydrogen 1s level in the respective cluster, and the valence electrons of Li_n clusters are likely to transfer from the Li 2s level to the H 1s level with consequent lowering in the E_H value (see Table I). When hydrogen-induced levels are sufficient enough to occupy all valence electrons of clusters, AIP and VIP show a sudden jump; i.e., IP becomes maximum for saturated lithium Li_nH_n clusters. This is true for all cluster sizes. A study by Chandrakumar et al.³⁹ has revealed that the cube root of polarizability per atom $\alpha^{1/3}/n$ is inversely proportional to the inverse of ionization energy per atom VIP^{-1}/n for both lithium Li_n and sodium Na_n clusters for $n = 2-10$. To explore the correlation between polarizability and ionization energy, the variation of $\alpha^{1/3}/n$ with VIP^{-1}/n is investigated for Li_8H_m and displayed in Figure 4.

A strong correlation is observed in which $\alpha^{1/3}/n$ increases almost linearly with respect to VIP^{-1}/n . This correlation implies that lower ionization energy leads to a stronger binding of the valence electrons and in turn leads to a higher value of static dipole polarizability which is a measure of distortion of electron density under an external electric field.

III. Conclusion. The current work presents the exhaustive study of Li_nH_m clusters up to a large size. The sequential addition of hydrogen to lithium metal Li_n clusters has been studied by the DFT method under a B3LYP hybrid correlational functional. The accuracy of the method employed has been established by

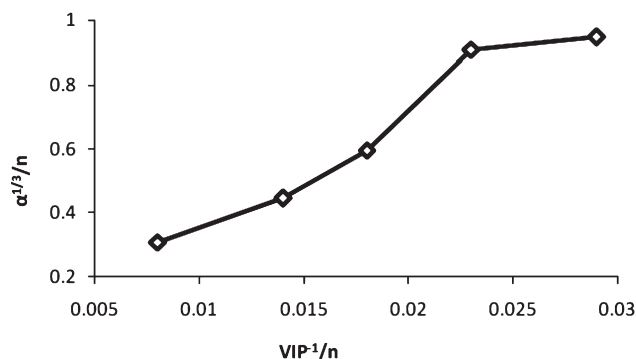


Figure 4. Cube root of polarizability per atom $\alpha^{1/3}/n$ vs inverse of ionization energy per atom VIP^{-1}/n . For Li_8H_m cluster.

excellent agreement with the available literature and experimental data. The results are analyzed as a function of cluster size and composition. In particular, we follow the evolution of the properties as the composition changes from metal-rich ($m < n$) to stoichiometric ($m = n$). The sequential addition of hydrogen to small-sized Li clusters led to the conjecture that Li_nH_m clusters ($n + m = \text{even number}$) are more stable. This has been attributed to their closed shell configuration that makes it difficult to remove an electron from the doubly occupied HOMO level. There is a significant increase in the IP of the completely hydrogenated, i.e., Li_nH_n clusters for all sizes. This result is consistent with the conclusion drawn by Fuentealba et al.²² in the analysis of small-sized Li_4H_4 and Li_9H_9 clusters that the Li_nH_n clusters have the bonds modeling an insulator. The present study demonstrates that polarizability is related to the cluster geometry, and α decreases as clusters become more compact with successive addition of hydrogen. As per the conjecture of perturbation theory, α is found to be inversely related to the HOMO–LUMO gap. The inverse correlation between the cube root of polarizability per atom $\alpha^{1/3}/n$ and the vertical ionization energy per atom VIP^{-1}/n that holds good for Li_n clusters³⁹ works for Li_nH_m clusters as well. It has been established that the most stable structure of Li_4H_4 has T_d symmetry, while Li_6H_6 is a hexagonal unit and the Li_9H_9 structure resembles fcc lattice. As the cluster is hydrogenated, it tends to form regular lattice regions with alternating Li and H ions. $\text{Li}_{18}\text{H}_{18}$, $\text{Li}_{20}\text{H}_{20}$, and $\text{Li}_{30}\text{H}_{30}$ studied by us are being reported for the first time. Interestingly, the fcc structure of the bulk LiH crystal appeared at the sizes investigated in the present work, and the lattice constant gradually approached the bulk value. In general, the metal hydrogen interaction in the Li_nH_m cluster is qualitatively similar to that in the bulk LiH crystal. $\text{Li}_{13}\text{H}_{14}$ is reportedly studied here for the first time to see if it is possible to add hydrogen in excess of the metal atom. It has been found that the excess hydrogen is present at the interstitial space, indicating that it is possible to accommodate more hydrogen in the lithium hydride cluster than the corresponding bulk. However, further investigations are required to reach a firm conclusion. The vibrational frequencies computed for Li_nH_m clusters showed no regular trend as cluster size increases, and therefore no general comment over the trend of vibrational frequencies with increase of cluster size can be made.

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