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Plasma-Assisted Nanoscale Protein Patterning on Si Substrates via Colloidal Lithography

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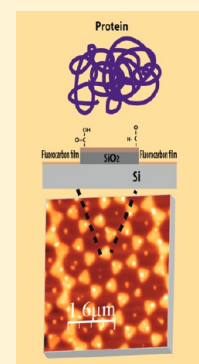
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Supporting Information

ABSTRACT: Selective immobilization of proteins in well-defined patterns on substrates has recently attracted considerable attention as an enabling technology for applications ranging from biosensors and BioMEMS to tissue engineering. In this work, a method is reported for low-cost, large scale and high throughput, selective immobilization of proteins on nanopatterned Si, based on colloidal lithography and plasma processing to define the areas (<300 nm) where proteins are selectively immobilized. A close-packed monolayer of PS microparticles is deposited on oxidized Si and, either after microparticle size reduction or alternatively after metal deposition through the PS close-packed monolayer, is used as etching mask to define SiO₂ nanoislands (on Si). C₄F₈ plasma was used to selectively etch and modify the SiO₂ nanoislands while depositing a fluorocarbon layer on the Si surface. The plasma-treated surfaces were chemically characterized in terms of functional group identification through XPS analysis and reaction with specific molecules. Highly selective protein immobilization mainly through physical adsorption on SiO₂ nanoislands and not on surrounding Si was observed after C₄F₈ plasma-induced chemical modification of the substrate. The thickness of the immobilized protein monolayer was estimated by means of AFM image analysis. The method reported herein constitutes a cost-efficient route toward rapid, large surface, and high-density patterning of biomolecules on solid supports that can be easily applied in BioMEMS or microanalytical systems.



1. INTRODUCTION

The potential of proteins to be immobilized on micro- or nanofabricated devices is steadily gaining importance for application in biosensors, BioMEMS, protein arrays, as well as tissue engineering.^{1–7} Protein patterning involves immobilization of biomolecules in particular areas on solid surfaces while preventing protein attachment on the rest of the surface.^{8,9} The desired feature size depends on the application, and the advancement of technology from micro- to nanofabrication has favored protein patterning shift from the microscale to the nanoscale for applications ranging from biosensors and protein arrays to basic studies of single protein interactions and cell attachment.^{10–14} To create protein patterns, photolithography-based methods have been extensively used.⁸ However, for features smaller than 1 μm, photolithography using contact or proximity printing is reaching its limits and becomes non cost-efficient. On the other hand, dip-pen lithography (using AFM tips) and e-beam lithography that can successfully achieve nanoscale patterning involve complex and high-cost apparatus, are serial processes and, as such, lack scalability.^{8,15}

A versatile, easy, fast, flexible, and low-cost lithographic method to reach successfully the nanometer scale is colloidal lithography. Its capability for high-resolution patterning renders it appropriate for nanoscale protein patterning in emerging biomedical applications.^{16,17}

Colloidal lithography is based on the close-packed self-assembly of colloidal microparticles over large surface areas. This method has been extensively used as a patterning technique,^{10,11,14,16,18–21} with the colloidal particles acting as masks in metal/polymer deposition and/or plasma etching.^{15,16,22–26} Moreover, colloidal microparticles are used for the enhancement of surface nanotopography.^{16,27,17}

In this work, we report a novel protein nanopatterning method using colloidal lithography^{11,16} and plasma-based^{12,28,29} chemical modification of SiO₂/Si substrates in C₄F₈²⁷ that

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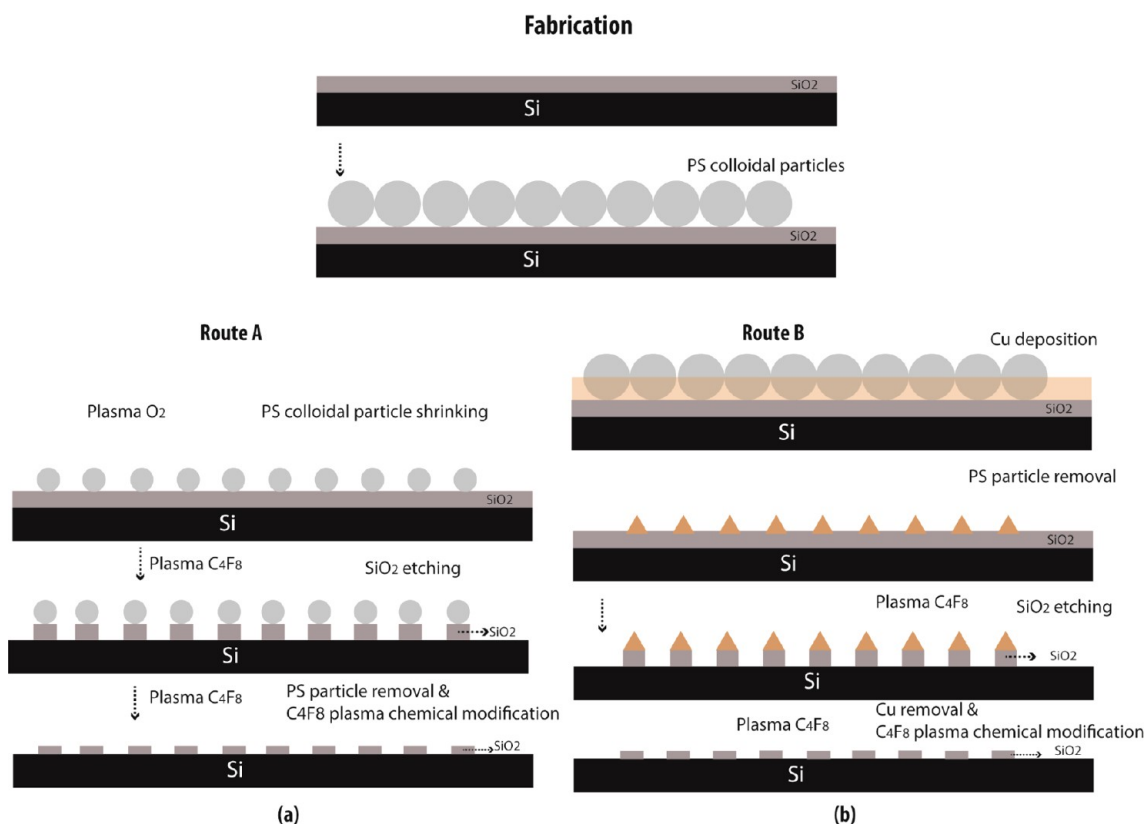


Figure 1. Schematic of the two processes for fabrication of SiO₂ nanoislands on Si substrates.

enables selective immobilization of proteins only on the plasma-modified SiO₂ nanoislands and not on the plasma-modified Si surface. The greatest advantage of our method is the simplicity of the procedure employed for selective chemical modification of the surface, i.e., short plasma treatment, as compared to functionalization protocols based on wet chemistries.^{7,13,15,16,23,24,26,30} We demonstrate SiO₂ nanoislands patterned on a Si substrate, using a two-dimensional (2D) closed-packed monolayer of colloidal particles after size reduction or alternatively deposited metal islands between the particles, as an etching mask for defining the SiO₂ nanoislands. Following an optimized plasma treatment, selective immobilization of proteins is achieved on the SiO₂ nanoislands with respect to Si.²⁸ The thickness of the selectively immobilized protein layer on SiO₂ is estimated by means of atomic force microscopy (AFM) and Matlab-based software for image analysis. Finally, to elucidate the chemical composition of our plasma-modified substrates, we combined XPS spectroscopy with identification of the major functional groups after reaction with specific biomolecules. The latter were selected to form covalent bonds with carboxyl and/or carbonyl groups. The confirmed presence of such functional groups on SiO₂ surfaces, as a result of selective plasma modification of our substrates, suggests the possibility of covalent protein binding on such surfaces apart from physical adsorption.

2. EXPERIMENTAL SECTION

2.1. Materials. Typical 3 in. silicon wafers with thermally grown oxides (with a thickness of 60 nm) were used as substrates after cleaning in piranha solution (H₂SO₄/H₂O₂ 1:1) to remove any possible contamination. (**Caution!** Piranha solution is aggressive and explosive. Never mix piranha waste

with solvents. Check the safety precautions before using it.) A suspension of 0.92 μm polystyrene (PS) particles in water were purchased from Mikropartikel GmbH. Triton X-100 was purchased from Sigma Aldrich. Plasma treatment was performed using O₂ and c-C₄F₈ gas provided by Air Liquide Hellas.

Bovine serum albumin (BSA, Cohn fraction V, RIA grade), D-biotin and biotinamido-hexanoic acid hydrazide were also purchased from Sigma Chemical Co. (St. Louis, MO, USA). Amine-PEG₃-biotin, EDC (1-ethyl-3-(3-(dimethylamino)-propyl)carbodiimide hydrochloride), and sulfo-NHS (N-hydroxysulfosuccinimide) were purchased from Thermo Fisher Scientific Inc. (Rockford, IL, USA). Streptavidin labeled with AlexaFluor 546 (AF546) was purchased from Molecular Probes, Inc. (Eugene, OR, USA).

2.2. Methods. **2.2.1. Colloidal Lithography.** Spin coating was used for the deposition of colloidal particles on oxidized Si substrates. A PS bead solution was prepared by mixing equal volumes of particle suspension in water with a 1:400 (v/v) Triton X-100/methanol solution. Triton-X is used to reduce the surface tension of the particle aqueous solution and thus increase the adhesion of the particles with the piranha-cleaned substrates. The coating process consisted of two steps: (a) a dispersion step at 200–300 rpm/min for approximately 30 s and (b) a second spinning step at 1000 rpm/min for 10–15 s, to remove the excess bead solution, as described in detail in our previous work.³¹

2.2.2. Plasma Processing. Plasma processing of the substrates took place in a high-density plasma reactor (Helicon plasma reactor, Micromachining Etching Tool, MET, from Alcatel).

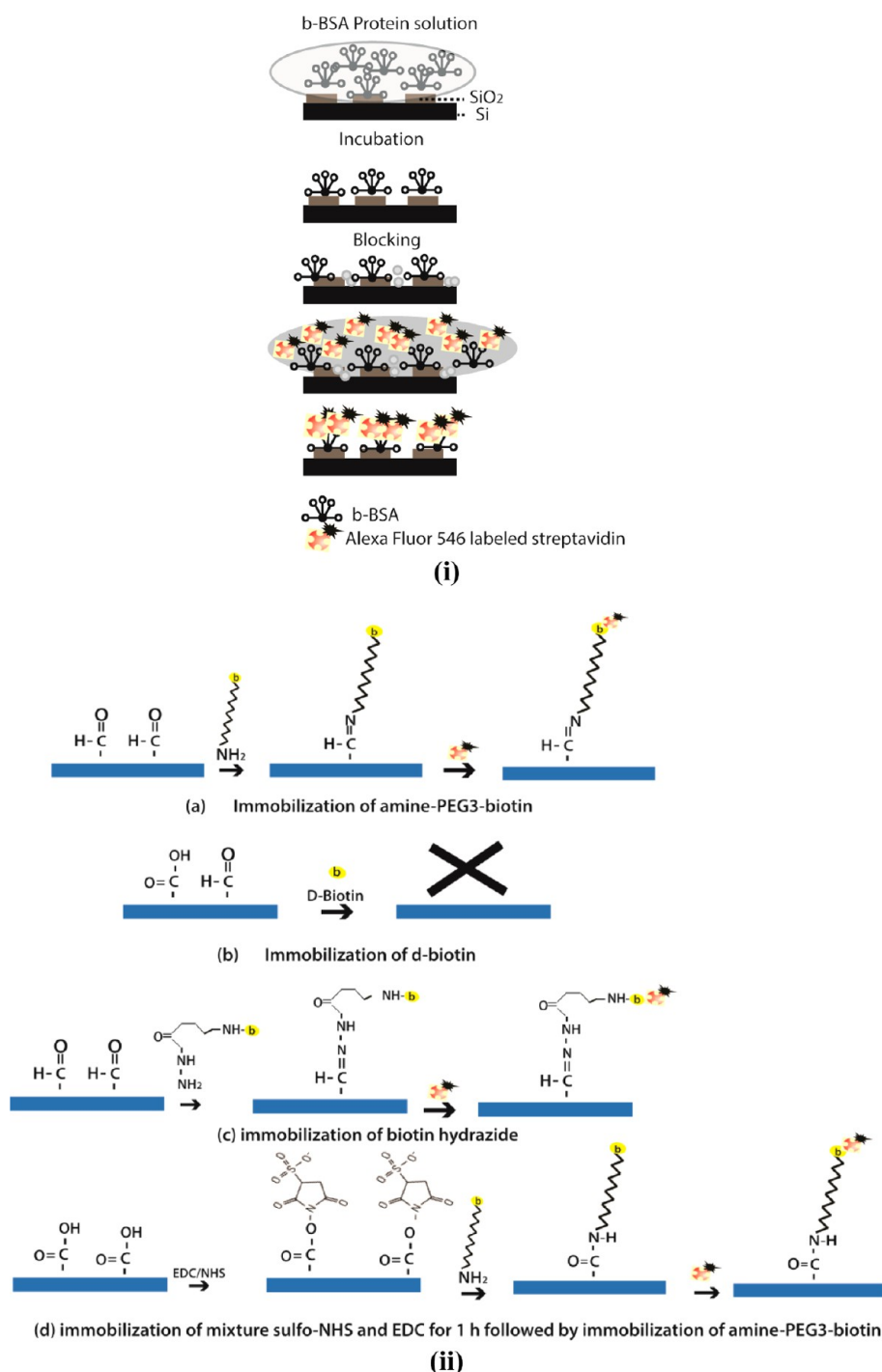


Figure 2. (i) Process flow for selective b-BSA protein immobilization on selectively modified patterned substrates. (ii) Processes of covalent attachment and detection of biotin derivatives on Si and SiO₂ surfaces.

For PS particle size reduction, isotropic O₂ plasma was used (gas pressure, 0.75 Pa; electrode temperature, 15 °C; O₂ flow rate, 100 sccm; plasma source power, 1900 W; bias voltage, 0 V).

For SiO₂ etching, between the PS microparticles or the copper nanoislands, as well as the selective chemical modification of the SiO₂ with respect to Si areas, a c-C₄F₈ plasma was used at the following conditions: gas pressure of 0.26 Pa, electrode temperature of 0 °C, C₄F₈ flow of 25 sccm, plasma source power of 800 W, electrode bias at −160 V.

2.2.3. Copper Deposition. Cu films with a total thickness of 28 nm were deposited by magnetron sputtering at room temperature using a MANTIS deposition system from a Cu (2 in. disk) target. Prior to the deposition, the chamber was evacuated to a base pressure of 7×10^{-7} Torr and the process gas (Ar 5N) pressure was 2.5 mTorr. A deposition rate of 5.1 ± 0.1 nm/min was achieved by applying 60W RF.

2.2.4. Substrate Patterning. After PS particle spin-coating on SiO₂, a closely packed particle assembly is produced. Two different routes are then employed to create ordered arrays of SiO₂ nanoislands (Figure 1).

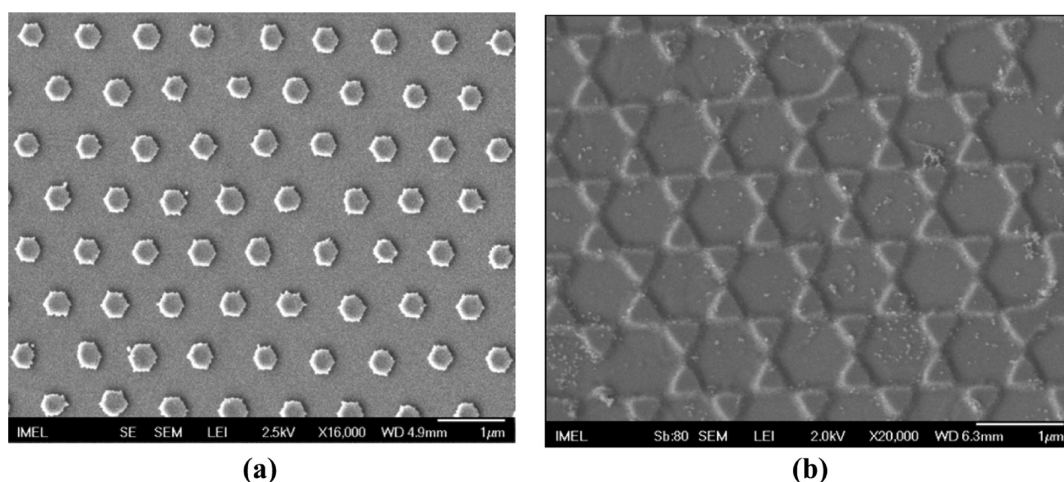


Figure 3. SEM images of our patterned SiO₂/Si substrates. SiO₂ nanoislands of (a) circular shape, 300 nm in diameter, fabricated following route A, and (b) triangular shape, with a side of 300 nm, fabricated following route B, are depicted.

According to the first route (route A), the substrate patterning process consists of a two-step plasma etching. First, an isotropic etching step in O₂ plasma takes place in order to shrink the particles followed by an anisotropic plasma treatment step in C₄F₈ to completely remove the 60 nm thick SiO₂ layer from the areas between the particles. This allows the fabrication of SiO₂ nanoislands of controlled diameter depending on the etching time and the applied bias voltage. Subsequently, the PS particles are removed from the top of the SiO₂ nanoislands using a strong adhesive tape (Figure 1a). The second route (route B) involves copper deposition in the interstices between the particles right after spin-coating of colloidal particle solution on SiO₂/Si substrates (Figure 1b). PS particles are removed and the copper remained islands are utilized as etch mask during an anisotropic plasma etching step in C₄F₈, which completely removes the 60 nm thick SiO₂ layer from the areas between the copper islands. After that step, copper is removed and the SiO₂ nanoislands are revealed on Si. For copper removal, a commercial PCB copper etchant, sodium persulfate, was used (100 gr of powder diluted in 500 mL water at 50 °C).

Finally, in both routes, an optimized treatment step in c-C₄F₈ plasma is performed for selective chemical modification of the SiO₂ nanoislands with respect to Si areas.²⁸

2.2.5. Protein Deposition. A model binding assay based on the biotin–streptavidin system was employed to demonstrate selective protein immobilization to freshly prepared, typically 1 day after C₄F₈ selective plasma treatment, nanopatterned surfaces. In particular, biotinylated bovine serum albumin (b-BSA) was immobilized onto the surface (without any additional chemical treatment of the surface) and was detected through reaction with fluorescently labeled streptavidin. Two-microliter droplets of b-BSA solution (25 µg/mL) in 50 mM phosphate buffer, pH 7.4, were manually deposited onto the surface and incubated for 1 h at room temperature (RT) in a humidity chamber. After washing, the samples were immersed in a 10 g/L BSA solution in 50 mM phosphate buffer, pH 7.4 (blocking solution), for 1 h at RT, to cover the remaining free binding sites of the surface. The proteins immobilized onto the surface were detected through reaction with a 5 µg/mL solution of AF546-labeled streptavidin in blocking solution, for 30 min at RT. Subsequently, the surfaces were washed extensively with 50 mM phosphate buffer, pH 7.4, 0.05% Tween 20 (v/v), and

distilled water and dried under N₂ stream. The protein immobilization and detection process is depicted schematically in Figure 2. The protein adsorption on the patterned substrates was confirmed by fluorescence images acquired through a confocal microscope.

2.2.6. Covalent Binding of Biotin Derivatives. All processes followed for covalent binding of activated biotin derivatives onto the patterned and plasma-assisted functionalized surface as well as their detection are schematically depicted in Figure 2(ii). To identify the presence of carbonyl groups on the surfaces, the direct immobilization of (a) amine-PEG₃-biotin, (b) biotinamidohexanoic acid hydrazide, and (c) D-biotin was investigated. All three reagents were diluted at a concentration of 25 mM in 0.1 M NaHCO₃ solution, pH 8.5, and 10 µL droplets of each solution were deposited onto the surfaces. Incubation for 2 h at RT in a humidity chamber was implemented.

For the detection of carboxyl groups onto the surfaces, a two-step approach was followed: (1) activation of carboxyl groups by a EDC/NHS mixture (10 mM EDC and 5 mM sulfo-NHS in 0.1 M MES buffer, pH 5) for 1 h at RT. (2) Coupling of amine-PEG₃-biotin from a 25 mM solution in 0.1 M NaHCO₃ solution, pH 8.5, for 2 h at RT.

Then all samples were washed with 0.1 M NaHCO₃ solution, pH 8.5, and immersed in a 10 g/L BSA solution in 50 mM phosphate buffer, pH 7.4 (blocking solution), for 1 h at RT to cover the remaining free binding sites of the surface. After that, the surfaces were washed three times with 50 mM phosphate buffer, pH 7.4. The immobilized biotin derivatives on the surface were detected through reaction with a 2.5 µg/mL AF546-labeled streptavidin in blocking solution, for 30 min at RT. Subsequently, the surfaces were washed extensively five times with 50 mM phosphate buffer, pH 7.4, and distilled water and dried under a stream of N₂. Fluorescence images were acquired with an Axioskop 2 Plus epifluorescence microscope (Carl Zeiss) equipped with a Sony Cyber-Shot 8-bit digital camera and processed with ImagePro Plus software (Media Cybernetics, Inc.).

2.2.7. Surface Characterization. A JEOL JSM-7401F FEG SEM was used for observation of the surfaces after colloidal lithography and plasma treatment. Also, a confocal microscope from Leica Microsystems (TCS SP5) was used for observation of the surfaces after protein immobilization and reaction with

232 fluorescently labeled binding molecules, to demonstrate
233 selective protein adsorption on the SiO₂ nanoislands.

234 The surface topography was characterized by an atomic force
235 microscope (AFM, Veeco diInnova) in the tapping mode. The
236 scanning rate was 1 kHz, and the scanned area $4 \times 4 \mu\text{m}^2$ with
237 512×512 pixel resolution. After a second-order polynomial
238 plane correction, the root-mean-square (rms) surface roughness
239 was calculated from the software of the instrument.

240 The chemical composition of the surfaces prior to and after
241 plasma treatment was investigated by XPS analysis. The spectra
242 were recorded on a PHI 5000 VersaProbe II (ULVAC-PHI,
243 Chigasaki, Japan) system using a microfocused ($100 \mu\text{m}$, 25 W)
244 Al K α X-ray beam with a photoelectron takeoff angle of 45°. A
245 dual-beam charge neutralizer was used to compensate the
246 charge-up effect. High-resolution spectra were collected with
247 analyzer pass energy of 23.5 eV. The operating pressure in the
248 analytical chamber was less than 5×10^{-7} Pa. All XPS peaks
249 were referenced to the neutral (C–C) carbon C1s peak, at a
250 binding energy of 284.8 eV. Spectral backgrounds were
251 subtracted using the Shirley method.

3. RESULTS AND DISCUSSION

252 **3.1. Patterned Substrates.** Following the above-described
253 method of surface nanopatterning, Scanning electron micros-
254 copy (SEM) was performed, to evaluate our nanopatterned
255 substrates before protein deposition. The images in Figure 3a,b
256 show the formation of SiO₂ nanoislands, of circular or
257 triangular shape (depending on the fabrication route) with a
258 maximum dimension of 300 nm. It should be pointed out that
259 although the final SiO₂ feature size obtained through PS
260 microparticle shrinking (route A) can be as low as possible
261 depending on the etching time of the shrinking step, high
262 particle size variability was observed at long etching times;
263 therefore, shorter etching durations were preferred, i.e., 130 s
264 etching, leading to microparticle shrinking to about one-third of
265 their initial size (Figure 3a). The size of the triangular-like SiO₂
266 islands shown in Figure 3b is defined by the areas left between
267 the 2D closed-packed PS microparticle arrays, and each
268 equilateral side is approximately 300 nm.

269 **3.2. Chemical Analysis of Si and SiO₂ Surfaces.** After
270 nanopatterning, our substrates undergo a short (15 s) C₄F₈
271 plasma step, optimized such that although a fluorocarbon layer
272 is deposited on Si areas, SiO₂ areas remain hydrophilic (water
273 contact angle of 60°). We have shown that on micropatterned
274 substrates²⁸ this step leads to a distinct chemical modification
275 of SiO₂ with respect to Si areas, which in turn leads to selective
276 immobilization of proteins on SiO₂ areas mainly by physical
277 adsorption, with no other functionalization being necessary.
278 The overall atomic composition of the Si and SiO₂ surfaces
279 before and after plasma treatment is shown in Table 1. As
280 expected, the overall C atomic concentration in both plasma-
281 modified substrates was increased, in Si from 8.9% to 27% and

in SiO₂ from 3.8% to 10.5%. More importantly, the modified Si
surfaces show a significant increase of the F atomic
concentration from ~0% to ~25.2% after 15 s treatment. On
the other hand, the F atomic concentration of the modified
SiO₂ surfaces increased less (from ~0% to ~7.0%) compared to
Si surfaces.

To investigate the plasma-induced chemical modification of
our substrates, further XPS analysis was performed before and
after plasma treatment, followed by curve fitting analysis for
both the C1s and F1s spectra. The contribution of all
components and curve fitting data of the C1s spectra for Si
and SiO₂ before and after 15 s C₄F₈ plasma treatment are
shown in Table 2. The C1s spectra of untreated SiO₂ and Si
samples are characterized basically by the presence of C–C and
C–O moieties, typical for native Si and SiO₂ surfaces.
Additional components are introduced in both Si and SiO₂
samples from the plasma modification step, as it is suggested by
the XPS data. Analysis revealed additional contributions from
–[CF₂]_n– (287.6 eV), COOH or CF (289 eV), CF₂ (291.3
eV), and CF₃ (293.6 eV) groups, whereas the contribution of
the C–O or CF_x–CH₂ (286.4 eV) component is greatly
enhanced. This result shows that a Teflon-like film in –(CF_n)–
form is deposited selectively on Si substrate after C₄F₈ plasma,
confirming our previous results.²⁸ In addition, careful
examination of Table 2 shows that peaks like (C–O) CF_x–
CH₂ and COOH or CF appeared to a higher proportion after
plasma modification on SiO₂ compared to Si. The presence of
COOH and C=O is proved below through reaction with
appropriately functionalized biotin derivatives.

3.3. Detection of Specific Functional Groups for Covalent Binding. To determine the presence of carboxyl
and/or carbonyl groups (that could be used for covalent
immobilization of biomolecules) on the plasma-modified SiO₂
and Si surfaces compared to untreated ones, we used biotin
derivatives that are designed to form covalent bonds with
common functional groups like carboxyl (–COOH) or
carbonyl (C=O). The reagents used were (1) amine-PEG₃-
Biotin, (2) D-biotin, (3) biotinamidohehexanoic acid hydrazide,
and (4) a mixture of sulfo-NHS and EDC followed by amine-
PEG₃-biotin (see Experimental Section). C₄F₈ plasma treat-
ment forms C–C, C=O and CF_x–*CH₂, (CF₂)_n, COOH,
and CF groups onto plasma-modified SiO₂ and Si, as revealed
with XPS (see previous section), through which the above
biotin derivatives are covalently attached to the surfaces. More
specifically, the –NH₂ group of the amine-PEG₃-biotin and the
hydrazide group from biotinamidohehexanoic acid hydrazide
covalently links to carbonyl groups from the plasma-treated
substrate. Please note that hydrazides react more efficiently to
carbonyl groups, thus confirming the presence of these specific
groups onto the treated surfaces. Also the mixture sulfo-NHS
and EDC activates the carboxyl groups from the modified
substrate forming an amine-reactive O-acylisourea intermediate.
This intermediate reacts then readily with the –NH₂ group of
the amine-PEG₃-biotin, yielding a stable amide bond. D-biotin
was used as a control biomolecule to exclude binding of the
derivatives from sites other than their specific functional
groups, because it has not a functional group that can covalently
bind to carboxyl or carbonyl groups. In Figure 4a,b, the
fluorescence intensities obtained from areas modified with the
respective biotin derivatives are provided for both untreated
and 15 s C₄F₈ plasma-treated Si and SiO₂ surfaces, after
reaction with AF546-labeled streptavidin. The solid bars 343

Table 1. Composition (Atomic Element Concentration) of Si and SiO₂ Surfaces before and after Plasma Modification

At (%)	Si untreated	Si C ₄ F ₈ -treated	SiO ₂ untreated	SiO ₂ C ₄ F ₈ -treated
C	8.9	26.7	3.8	10.5
Si	49.8	28.9	30.3	26.2
O	41.4	17.8	65.9	55.7
F	0.0	25.2	0.0	7.0
N	0.0	1.4	0.0	0.6

Table 2. Relative Percentage of Different Components in the C1s Peak for Si and SiO₂^a

binding energy (eV)	284.8	286.4	287.7	289	291.3	293.5
moieties	(C–C) CH ₂	(C–O) CF _x –CH ₂	H replaced by F in –[CH ₂] _n –	COOH and CF	[^] CF ₃ –[CF–*CF ₂] _n – [^] CF ₃ *CF ₂	
Si untreated	76.5	19.6		4		
Si treated	37.9	18.6	13.4	11.6	5.0	13.5
SiO ₂ untreated	85.9	14.6				
SiO ₂ treated	42	19.6	6.2	16.3	3.9	12.1

^aTypical binding energies are given at ±0.1 eV.

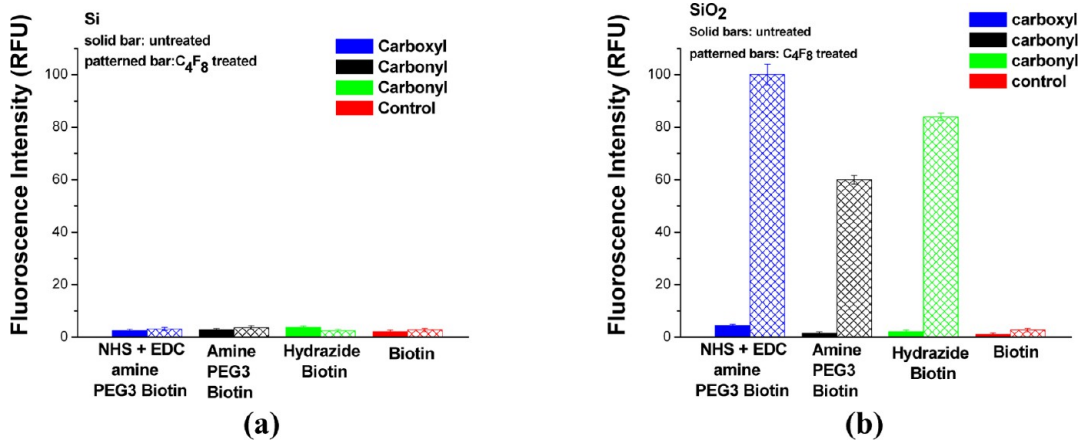


Figure 4. Fluorescence intensities obtained from areas modified with biotin derivatives: (1) amine-PEG₃-biotin (black bars), (2) sulfo-NHS followed by amine-PEG₃-biotin (blue bars), (3) D-biotin (red bars), and (4) biotin hydrazide (green bars) are provided after 2 h incubation and after reaction with AF546-labeled streptavidin on both untreated and 15 s C₄F₈ plasma-treated Si and SiO₂ surfaces.

represent the untreated substrates, whereas the patterned bars represent the treated substrates.

As shown in Figure 4a, both treated and untreated Si substrates exhibit fluorescence absolute values comparable to the untreated SiO₂ values. On the other hand, SiO₂ plasma-modified surfaces provided high fluorescence values, indicating reaction of carboxylic groups with NHS/EDC and/or amine biotin or of carbonyl groups with amine-PEG₃-biotin and biotin hydrazide. More specifically, the fluorescence values obtained by reacting the respective biotin derivative with surface carboxylic groups was 36-times higher than the value obtained by applying D-Biotin onto the same surface (control). Furthermore, the reaction of the surfaces with amine-PEG₃-biotin and hydrazide biotin provided 21- and 30-times higher fluorescence, respectively, compared with the control untreated surface.

These results suggest that C₄F₈ plasma treatment of SiO₂ leads to the formation of reactive and available carboxylic and carbonyl groups that can be used for covalent immobilization of biomolecules. Although the *potential* of such plasma-treated surfaces for covalent binding of biomolecules is indicated here, proteins are selectively immobilized on our plasma-modified SiO₂ surfaces mainly through physical adsorption. Nevertheless, covalent binding of proteins is also possible due to the existence of carboxyl and carbonyl groups on such surfaces, provided that functionalization of these groups by appropriate cross-linkers (e.g., EDC/NHS) will be first performed. We suspect that the presence of such groups selectively on the C₄F₈ plasma-treated SiO₂ surfaces (and not on Si surfaces) is a result of the reaction of plasma-deposited carbon with oxygen present on SiO₂ surface, as well as of aging when exposed to atmosphere after

plasma treatment. However, we are convinced that our substrates are stable, because experiments performed several weeks after sample preparation did not exhibit any sign of deterioration of the selectivity between plasma-treated SiO₂ and Si surfaces.

3.4. Quantification of Protein Immobilization on SiO₂ Nanoislands. The selective binding of proteins onto C₄F₈ plasma-treated, nanopatterned substrates was first determined through fluorescence images taken after streptavidin specific binding to b-BSA by means of confocal microscopy. The latter provided a clear evidence that protein immobilization occurred selectively on SiO₂ nanoislands. However, due to the limited resolution of the method to depict the obtained nanoislands (300 nm), the reliable estimation of the achieved selectivity (i.e., spot to background fluorescence intensity ratio) was not possible by confocal microscopy. Thus, an alternative technique was selected on the basis of AFM imaging for the estimation of the thickness of the protein layer on the SiO₂ nano-patterns.^{21,23,32} In fact, AFM imaging is widely used in the literature for protein thickness evaluation.²³ According to this approach, AFM images were taken before and after protein deposition and the height histograms of the measured surface topographies are plotted. They are expected to exhibit two peaks, the first one corresponding to the background (Si) surface, whereas the second peak corresponding to the nanoisland (SiO₂) surface. The distance between the peaks represents the nanoisland height in the image. After protein immobilization on the SiO₂ nanoislands, the second peak is expected to move to higher height values, and therefore the distance between the peaks is expected to increase. The amount of the increase in the peak difference gives an average

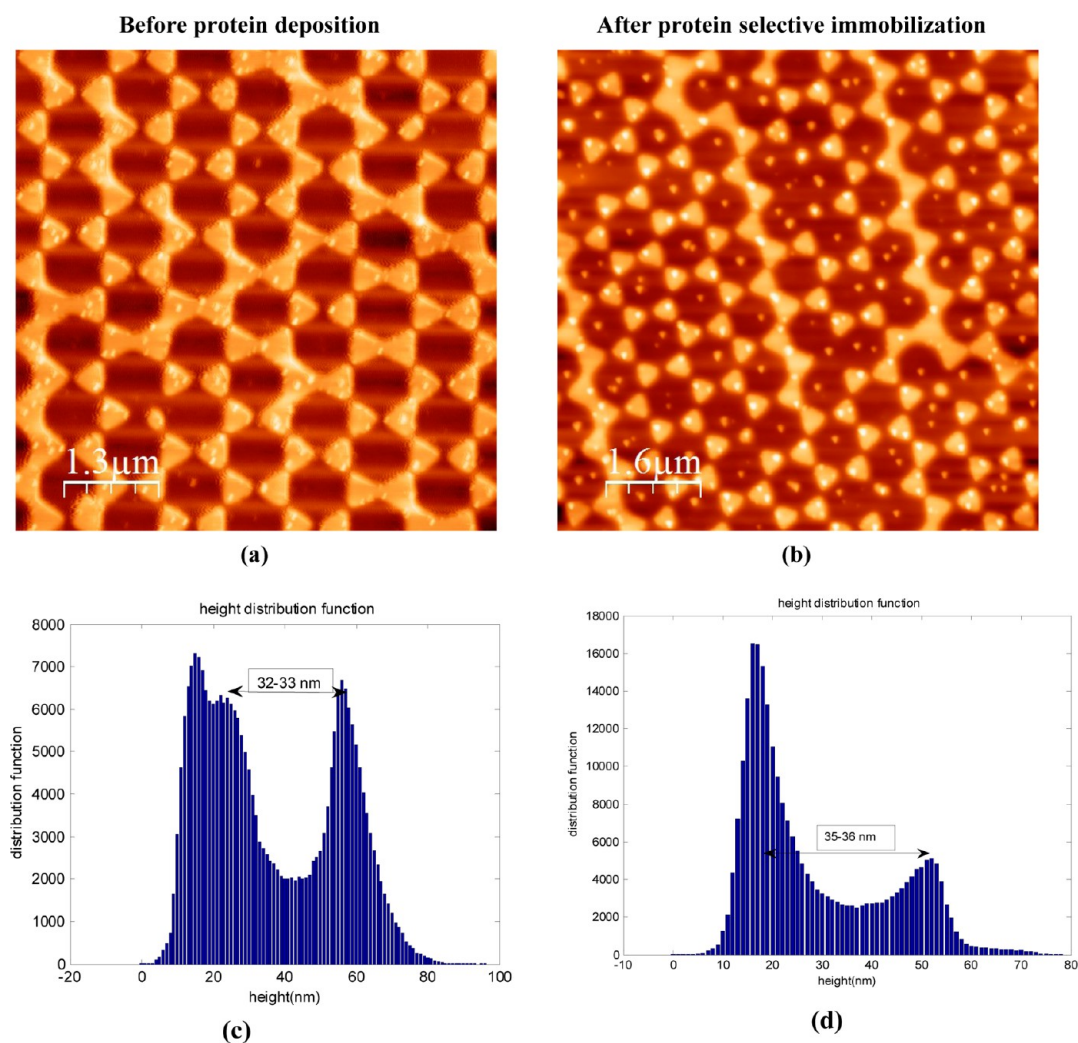


Figure 5. AFM images (a) before and (b) after b-BSA deposition, and (c), (d) corresponding surface height histograms after image analysis with “surfanalysis” (bin size 1).

estimation of the thickness of the deposited protein layer on SiO₂ nanoislands.

We have applied this method to our nanopatterned substrates and the results are shown in Figure 5, which depicts AFM images taken on the same sample consisting of posts with a diameter of 300 nm (route B), before (Figure 5a, scanning area: 5 × 5 μm²) and after b-BSA deposition (Figure 5b, scanning area: 8 × 8 μm²), and the respective height histograms (Figure 5c and Figure 5d) obtained after image analysis with the “surfanalysis” software developed in house.

In Figure 5c, the first peak which corresponds to the Si surface is composed of two peaks (Figure 5c), whereas this was not observed after protein immobilization (Figure 5d). This peak splitting is an artifact attributed to the color shading shown in Figure 5a (dark areas) for the Si surface, due to the “flying tip effect”.³³ This artifact is quite usual and appears in AFM images when the tip scanning direction coincides with the post orientation. After it was realized that the first peak of the background surface height distribution was an artifact, the second peak was taken into account as the actual background peak. According to this, the histogram of Figure 5c indicates that the average height of the SiO₂ nanoislands is approximately 32–33 nm, also independently confirmed by means of SEM measurement of the SiO₂ nanoisland height (after C₄F₈ plasma

treatment). Finally, it should be noted that the reason the “flying tip effect” was not observed in the images of Figure 5d (after protein immobilization) is due to the different AFM scanning orientation.

To increase the reliability of our measurements, we performed four AFM measurements, before and after protein immobilization, from which the SiO₂ height mean values were calculated and shown in Figure 6.

According to Figure 6, the mean value of the SiO₂ nanoisland height before protein deposition was 32 ± 0.5 nm, whereas the one corresponding to the nanoislands after protein immobilization was 35 ± 0.7 nm, suggesting a mean thickness of 3 ± 1 nm for the protein layer. This value is consistent with a BSA monolayer thickness (in dehydrated state) which is equal to 1.5–2 nm.^{34,35} This finding further confirms the selective immobilization of proteins on our plasma-modified SiO₂ nanoislands. As discussed in the previous subsection (Detection of Specific Functional Groups for Covalent Binding), it is assumed that these proteins are mainly physisorbed on our plasma-modified SiO₂ surfaces, although covalent binding through available on the surface reactive groups cannot be completely excluded. Polymers subjected to ion-assisted plasma processes have been shown^{29,36} to covalently immobilize biomolecules, due to reaction of biomolecules with surface

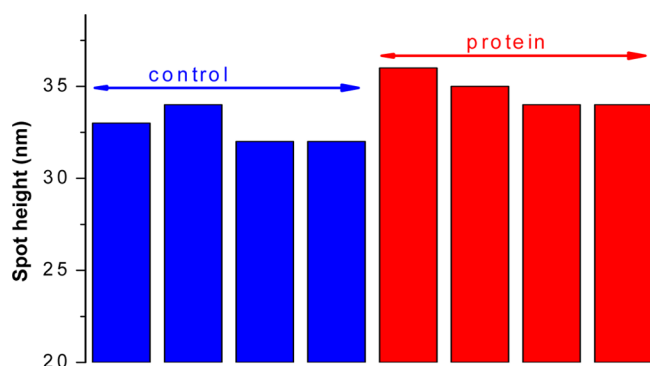


Figure 6. Nanoisland height distribution (four measurements) of samples before (blue bars) and after protein deposition (red bars).

free radicals. Although the possibility of the existence of free radicals on the plasma-treated surfaces cannot be excluded, such radicals are not expected on surfaces for ion implantation voltages less than 400 V,²⁹ as it is in our case (bias voltage of 100 V). Therefore, the potential of our plasma-treated SiO₂ surfaces to covalently bind biomolecules can be attributed more to the presence of carboxyl or carbonyl groups on the surface than to the presence of free radicals. In addition, no deterioration of the biomolecule immobilization capability of the plasma-treated SiO₂ surfaces was observed with aging, as it would be expected in the case of free radicals.

4. CONCLUSIONS

In this work, a novel, rapid and cost-efficient method for high-density protein nanopatterning appropriate for the modification on large surfaces is demonstrated on Si substrates. The method is based on colloidal lithography combined with plasma-induced surface modification to enable selective protein immobilization on C₄F₈ plasma-treated SiO₂ nanoislands and not on the surrounding Si surface. The thickness of the selectively immobilized protein monolayer on SiO₂ was estimated after height distribution analysis of AFM images before and after protein deposition. Extensive chemical characterization of the surfaces was performed combining XPS analysis and reaction with biomolecules that form covalent bonds with carboxyl or carbonyl groups, to detect the functional groups that are formed on the C₄F₈ plasma-treated surfaces. The confirmed presence of such functional groups on the surface of SiO₂ surfaces, as a result of selective plasma modification of our substrates, suggests the possibility of covalent protein binding on C₄F₈ plasma-modified SiO₂ surfaces apart from physical adsorption. Such fabricated protein nanopatterned substrates can be applied in biosensors for enhanced detection sensitivity, as model substrates for protein–protein interaction studies, as well as substrates for cell cultivation and tissue engineering.

■ ASSOCIATED CONTENT

Supporting Information

High-resolution XPS C_{1s} spectra (and their deconvolution) for untreated and C₄F₈ plasma-treated Si and SiO₂ surfaces. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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Notes

The authors declare no competing financial interest.

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