

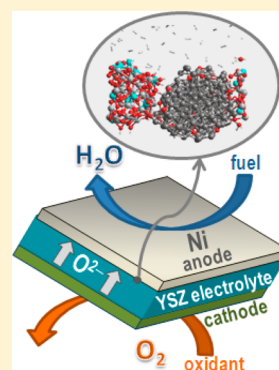
## ReaxFF Reactive Force-Field Modeling of the Triple-Phase Boundary in a Solid Oxide Fuel Cell

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## S Supporting Information

**ABSTRACT:** In our study, the Ni/YSZ ReaxFF reactive force field was developed by combining the YSZ and Ni/C/H descriptions. ReaxFF reactive molecular dynamics (RMD) were applied to model chemical reactions, diffusion, and other physicochemical processes at the fuel/Ni/YSZ interface. The ReaxFF RMD simulations were performed on the H<sub>2</sub>/Ni/YSZ and C<sub>4</sub>H<sub>10</sub>/Ni/YSZ triple-phase boundary (TPB) systems at 1250 and 2000 K, respectively. The simulations indicate amorphization of the Ni surface, partial decohesion (delamination) at the interface, and coking, which have indeed all been observed experimentally. They also allowed us to derive the mechanism of the butane conversion at the Ni/YSZ interface. Many steps of this mechanism are similar to the pyrolysis of butane. The products obtained in our simulations are the same as those in experiment, which indicates that the developed ReaxFF potential properly describes complex physicochemical processes, such as the oxide-ion diffusion, fuel conversion, water formation reaction, coking, and delamination, occurring at the TPB and can be recommended for further computational studies of the fuel/electrode/electrolyte interfaces in a SOFC.

**SECTION:** Surfaces, Interfaces, Porous Materials, and Catalysis



Solid oxide fuel cells (SOFCs) are among the most efficient fuel cells with theoretical efficiencies as high as 85%. Oxide-ion-conducting yttria-stabilized zirconia (YSZ) ceramics are widely applied as electrolytes in SOFCs, while perovskite-type oxides with good electronic and ionic conductivities and Ni/YSZ cermet usually serve as the cathode and anode, respectively. SOFCs operate at very high temperature ( $\sim 1000$  °C) to achieve sufficiently high oxygen conductivity, which creates some advantages and disadvantages. One of the important advantages is that SOFCs do not require precious metal catalysts, making them cheaper than other types of fuel cells. Another advantage is that SOFCs can be powered with a variety of fuels, including hydrogen, carbon monoxide, hydrocarbons, and combinations thereof.<sup>1,2</sup> In principle, the high operating temperature enables internal reforming reactions of hydrocarbon fuels, which make SOFCs even more attractive. In important respects, Ni is a suitable catalytic material for hydrocarbon reforming in SOFCs because the desired reactions are readily catalyzed on its surface. Unfortunately, if hydrocarbon fuels or other carbon-containing fuels are used, a layer of carbon is usually built up on the Ni catalyst surface due to coke formation, which rapidly breaks down the efficiency of the SOFC catalyst.<sup>3</sup> One of the most promising ways to alleviate this problem is alloying Ni with other metals such as Fe.<sup>4</sup> Regardless of whether pure Ni or a Ni alloy is used, the different thermal expansion coefficients of the catalytic and electrolyte materials cause another problem, delamination, which occurs at the electrode/electrolyte interface. Computa-

tional modeling is able to contribute significantly toward overcoming these obstacles by elucidating the fundamental mechanisms at work in these undesirable processes, so that rational approaches can be used in developing strategies for overcoming them. Furthermore, the computational methods prove to be very useful for determining the mechanisms of the desired chemical reactions that occur at the interface, for instance, hydrocarbon fuel conversion.

In our study, ReaxFF reactive force-field molecular dynamics (MD) were applied to model chemical reactions, diffusion, and other physicochemical processes at the fuel/Ni/YSZ interface. We used a ReaxFF potential (see Supporting Information), which combines two earlier potentials developed and validated to describe the YSZ electrolyte<sup>5</sup> and hydrocarbon chemistry catalyzed by Ni,<sup>6,7</sup> respectively. In ref 5, we reported the development of the ReaxFF reactive force field for the YSZ oxide-ion conducting electrolyte. It is based on substantial data derived from QM calculations on clusters and periodic systems, such as relevant pure metallic and metal alloy phases, and various bulk and surface oxide systems, including different atomic configurations for the YSZ electrolyte. To validate the use of the ReaxFF developed, we applied it to predict the oxygen ion diffusion coefficient in YSZ as a function of temperature. These values are in excellent agreement with

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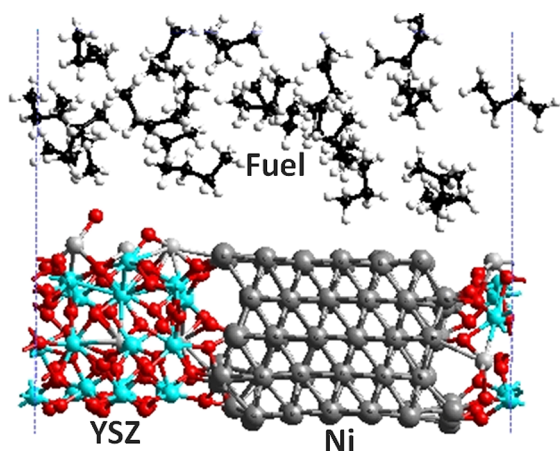
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experimental results,<sup>8,9</sup> setting the stage for the use of ReaxFF to model the transport of oxygen ions through the YSZ electrolyte for SOFCs. Because ReaxFF descriptions are already available for the Ni-metal catalyst,<sup>6</sup> we can now consider fully first-principles-based ReaxFF MD simulations of the critical functions in SOFCs, enabling the possibility of in silico optimization of these materials. The combined Ni/YSZ ReaxFF was developed by combining the YSZ and Ni/C/H descriptions and defining the missing angular terms (e.g., Ni–O–Zr) from combination rules. Ni–Zr and Ni–Y bonded interactions were ignored; Ni/Y and Ni/Zr nonbonded parameters were obtained from combination rules. Some examples of the ReaxFF versus QM energetics, which show the reliability of the force field, are available in the Supporting Information.

The ReaxFF MD simulations were performed on the H<sub>2</sub>/Ni/YSZ and C<sub>4</sub>H<sub>10</sub>/Ni/YSZ triple-phase boundary (TPB) systems at 1250 and 2000 K, respectively. The former temperature is close to the usual operating temperature of a SOFC, while the later temperature is approximately twice as high. The higher temperature was used to increase the rate at which the potential energy surface is sampled and the sampling of high energy configurations. Thus, while an elevated simulation temperature is necessary for the observation of reactive events within a computationally feasible simulation time, care must be taken in extending the conclusions drawn to systems at lower temperatures. Figure 1 shows the periodic system applied for

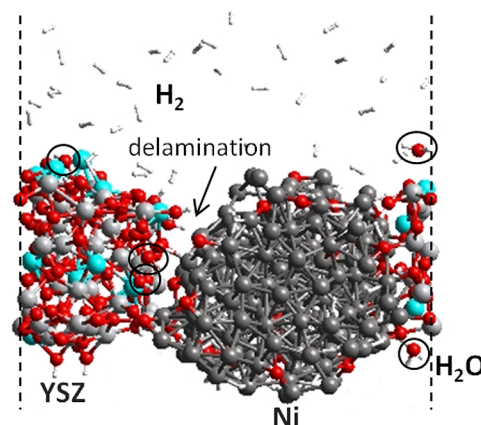


**Figure 1.** Periodic system used for modeling fuel conversion at the butane/Ni/YSZ interface.

modeling fuel conversion at the Ni/YSZ interface (240 Ni, 16 Y, 48 Zr, and 148 O atoms). We used the YSZ electrolyte with dopant concentration of ~14 mol % Y<sub>2</sub>O<sub>3</sub> [(Y<sub>2</sub>O<sub>3</sub>)(ZrO<sub>2</sub>)<sub>6</sub>]. The system was first equilibrated at 1250 and 2000 K using ReaxFF NPT dynamics and then we carried out ReaxFF NVT MD simulations at these temperatures on the corresponding TPB systems.

The number of fuel molecules and simulation temperature were varied depending on the problem studied to observe fuel conversion reactions within reasonable time period. The reaction barriers in the H<sub>2</sub>/Ni/YSZ system with 125 H<sub>2</sub> molecules enable us to observe numerous reactions within 150 ps of simulation time at the typical operating temperature, 1250 K, while the simulation temperature and time period were 2000 K and 2 ns for the C<sub>4</sub>H<sub>10</sub>/Ni/YSZ system with 20 butane molecules.

Figure 2 shows a snapshot of the H<sub>2</sub>/Ni/YSZ system following 150 ps of NVT dynamics at 1250 K, which now



**Figure 2.** Final point of the ReaxFF MD simulation on the H<sub>2</sub>/Ni/YSZ triple-phase boundary system.

includes 5 H<sub>2</sub>O, 51 H atoms on the Ni surface, 22 Ni–O bonds (Ni oxidation), and only 1 Ni–OH group. In addition, substantial reduction of the YSZ electrolyte, which results in the formation of 11 hydroxyl groups on the YSZ surface and 3 at the Ni/YSZ interface, was observed in our model system. Partially, the reduction of the YSZ surface might be due to the limited amount of diffusive oxygen available in the model that we used for the TPB.

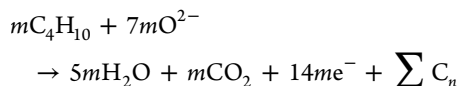
The anodic reaction in a SOFC is water formation:  $\text{H}_2 + \text{O}^{2-} \rightarrow \text{H}_2\text{O} + 2\text{e}^-$ . We identified two possible locations where water formation takes place in our simulation. The first is the Ni–YSZ interface, where (1) H<sub>2</sub> dissociates on Ni and forms Ni–H bonds, then (2) O<sup>2-</sup> comes from the YSZ electrolyte first to form OH, and finally (3) another H approaches OH to form H<sub>2</sub>O. The second location where we observe the water formation reaction is the Ni(111) surface. The first three steps at this location are similar to those previously described; however, once H<sub>2</sub>O has been formed it immediately leaves the Ni surface and moves to the YSZ surface, whereas the H<sub>2</sub>O molecules formed at the Ni/YSZ interface remain there for at least the duration of our simulation. The water formation mechanism observed in our simulation is in agreement with the mechanisms following from DFT studies<sup>10–12</sup> and from solving reaction–diffusion equations on both electrode and electrolyte surfaces.<sup>13,14</sup>

Our ReaxFF simulation not only unveils the individual mechanistic steps involved in water formation but also brings to light the structural changes that occur in each of the system's components in operando. As can be seen in Figure 2, the simulation indicates amorphization of the Ni surface and partial decohesion at the interface (so-called delamination), which have indeed all been observed experimentally.<sup>15</sup> In fact, the delamination of the Ni/YSZ interface in operando is well known, and one of the possible reasons is the difference in thermal expansion coefficients between Ni ( $13.3 \times 10^{-6}/\text{C}$ ) and YSZ ( $10 \times 10^{-6}/\text{C}$ ).<sup>15</sup> From the crystallographic viewpoint, the distinct atomic structures of Ni and YSZ may be responsible for decohesion at the interface as well. Regardless of the exact contributions of the thermal expansion coefficients and atomic crystal structures, the atomistic description given by our simulation is consistent with experimental the observations. Computational modeling may

help to better understand the mechanisms of electrode delamination in SOFCs, but this is beyond of the scope of our paper.

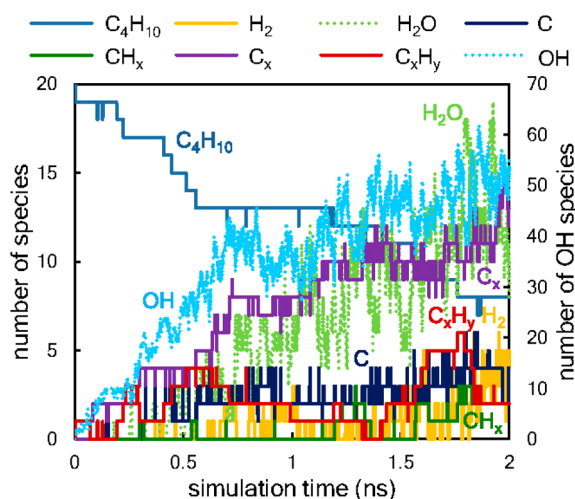
Having shown that our ReaxFF potential properly describes the water formation reaction at the Ni anode of a SOFC and thus allows first-principles-based predictions of the chemical processes at the H<sub>2</sub>/Ni/YSZ TPB, we now proceed to study a complicated hydrocarbon fuel, namely, butane.

The hydrocarbon-fuel-oxidation reactions, which deplete butane to produce water and carbon dioxide at the anode/electrolyte interface of a SOFC, can be put into the following general form



where the last term represents carbon deposits that cause coking of the anode. To simulate these chemical reactions at the butane/Ni/YSZ TPB, we replaced the H<sub>2</sub> fuel used in the previous simulations with 20 butane molecules. Because the reaction rate of butane is significantly slower than that of H<sub>2</sub>, we were forced to perform these simulations at an elevated temperature (2000 K) to observe significant reaction levels within a practical simulation time. At the end of our 2 ns simulation, we observed all major products—methane, ethane, ethylene, and propylene—found experimentally for the pyrolysis of *n*-butane.<sup>16–18</sup>

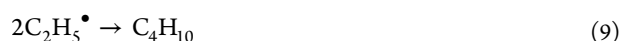
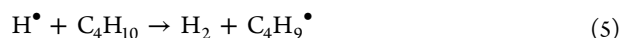
As can be seen in Figure 3, the number of butane molecules decreases dramatically during the first 500 ps before



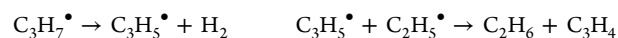
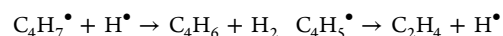
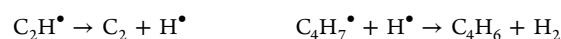
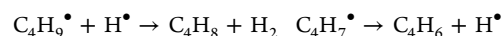
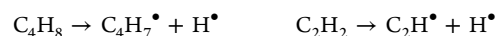
**Figure 3.** Populations of reactant, intermediate, and product species during 2 ns simulation of butane conversion at the Ni/YSZ interface.

temporarily leveling off, primarily due to the disappearance of empty surface sites and perhaps due to decreasing the number of butane molecules, which results in a lower fuel gas pressure. Indeed, fuel-decomposition reactions that take place over the entire course of the simulation show that primarily the adsorption, and not decomposition, of C<sub>4</sub>H<sub>10</sub> is inhibited. After 2 ns of NVT dynamics, 12 of the 20 C<sub>4</sub>H<sub>10</sub> have been consumed to form 10 H<sub>2</sub>O (4 in the gas phase), 54 OH (most of them on the YSZ surface), 18 H, 5 H<sub>2</sub> (all in the gas phase), 2 CH<sub>4</sub> (both gas phase), 1 C<sub>2</sub>H<sub>6</sub> (gas phase), 1 C<sub>2</sub>H<sub>4</sub> (gas phase), and 15 C<sub>n</sub> (from C to C<sub>8</sub>) species.

According to Purnell et al.,<sup>16,19</sup> the mechanism of the *n*-butane pyrolysis includes the following elementary reactions



In our simulation, we observe all of these reactions except for reactions 3 and 4. Instead, we observe reactions: CH<sub>3</sub><sup>•</sup> + C<sub>3</sub>H<sub>5</sub><sup>•</sup> → CH<sub>4</sub> + C<sub>3</sub>H<sub>4</sub> and C<sub>2</sub>H<sub>5</sub><sup>•</sup> + C<sub>3</sub>H<sub>5</sub><sup>•</sup> → C<sub>2</sub>H<sub>6</sub> + C<sub>3</sub>H<sub>4</sub>, in which similar products, CH<sub>4</sub> and C<sub>2</sub>H<sub>6</sub>, are formed. In addition, other species, such as C<sub>4</sub>H<sub>8</sub>, C<sub>4</sub>H<sub>7</sub>, C<sub>4</sub>H<sub>6</sub>, C<sub>3</sub>H<sub>5</sub>, C<sub>2</sub>H<sub>3</sub>, C<sub>2</sub>H<sub>2</sub>, C<sub>2</sub>H, CH<sub>4</sub>, CH<sub>3</sub>, CH<sub>2</sub>, H, C, C<sub>2</sub>, C<sub>3</sub>, C<sub>4</sub>, C<sub>5</sub>, C<sub>6</sub>, C<sub>7</sub>, OH, H<sub>2</sub>O, CH<sub>2</sub>O, CH<sub>3</sub>O, CH<sub>4</sub>O, C<sub>4</sub>H<sub>10</sub>O, and C<sub>4</sub>H<sub>9</sub>O, and reactions were found as well



and so on.

Many of these reactions are due to hydrogen abstraction, and some of the species are intermediates for the products shown in Scheme 1. In fact, in line with a previous study,<sup>20</sup> a tabulation of the frequency of each elementary reaction (see Supporting Information) suggests that hydrogen abstraction occurs significantly more readily and frequently than C–C bond cleavage, so that a sizable fraction of the adsorbed fuel is converted into carbon chains. Furthermore, we observe these carbon chains reacting with each other to form longer chains, which eventually clog the surface. This suggests that the preferential cleavage of C–H before C–C bonds and the facile C–C bond formation and cleavage in C chains on Ni are important factors in coking. Thus, modifying the metal catalyst by means of alloying, to increase the ease with which C–C bonds between partially hydrogenated carbon atoms take place





- (3) Clarke, S.; Dicks, A.; Pointon, K.; Smith, T.; Swann, A. Catalytic Aspects of the Steam Reforming Of Hydrocarbons in Internal Reforming Fuel Cells. *Catal. Today* **1997**, *38*, 411–423.
- (4) Hecht, E. S.; Gupta, G. K.; Zhu, H.; Dean, A. M.; Kee, R. J.; Maier, L.; Deutschmann, O. Methane Reforming Kinetics within a Ni-YSZ SOFC Anode Support. *Appl. Catal., A* **2005**, *295*, 40–51.
- (5) van Duin, A. C. T.; Merinov, B. V.; Jang, S. S.; Goddard, W. A., III. ReaxFF Reactive Force Field for Solid Oxide Fuel Cell Systems with Application to Oxygen Ion Transport in Yttria-Stabilized Zirconia. *J. Phys. Chem. A* **2008**, *112*, 3133–3140.
- (6) Mueller, J. E.; van Duin, A. C. T.; Goddard, W. A., III. Development And Validation of ReaxFF Reactive Force Field for Hydrocarbon Chemistry Catalyzed by Nickel. *J. Phys. Chem. C* **2010**, *114*, 4939–4949.
- (7) Nielson, K. D.; van Duin, A. C. T.; Oxgaard, J.; Deng, W. Q.; Goddard, W. A. Development of the ReaxFF Reactive Force Field for Describing Transition Metal Catalyzed Reactions, with Application to the Initial Stages of the Catalytic Formation of Carbon Nanotubes. *J. Phys. Chem. A* **2005**, *109*, 493–499.
- (8) Kilo, M.; Argirusis, C.; Borchardt, G.; Jackson, R. Oxygen Diffusion in Yttria Stabilised Zirconia - Experimental Results and Molecular Dynamics Calculations. *Phys. Chem. Chem. Phys.* **2003**, *5*, 2219–2224.
- (9) Minh, N. Q. Ceramic Fuel Cells. *J. Am. Ceram. Soc.* **1993**, *76*, 563–588.
- (10) Cucinotta, C. S.; Bernasconi, M.; Parrinello, M. Hydrogen Oxidation Reaction at the Ni/YSZ Anode of Solid Oxide Fuel Cells from First Principles. *Phys. Rev. Lett.* **2011**, *107*, 206103.
- (11) Shishkin, M.; Ziegler, T. Direct Modeling of the Electrochemistry in the Three-Phase Boundary of Solid Oxide Fuel Cell Anodes by Density Functional Theory: A Critical Overview. *Phys. Chem. Chem. Phys.* **2014**, *16*, 1798–1808.
- (12) Ammal, S. C.; Heyden, A. Combined DFT and Microkinetic Modeling Study of Hydrogen Oxidation at the Ni/YSZ Anode of Solid Oxide Fuel Cells. *J. Phys. Chem. Lett.* **2012**, *3*, 2767–2772.
- (13) Goodwin, D. G.; Zhu, H.; Colclasure, A. M.; Kee, R. G. Modeling Electrochemical Oxidation of Hydrogen on Ni–YSZ Pattern Anodes. *J. Electrochem. Soc.* **2009**, *156*, B1004–B1021.
- (14) Yao, W.; Croiset, E. Modelling and Ni/Yttria-Stabilized-Zirconia Pattern Anode Experimental Validation of a New Charge Transfer Reactions Mechanism for Hydrogen Electrochemical Oxidation on Solid Oxide Fuel Cell Anodes. *J. Power Sources* **2014**, *248*, 777–788.
- (15) Minh, N. Q. *Science and Technology of Ceramic Fuel Cells*; Elsevier: Amsterdam, 1995.
- (16) Purnell, J. H.; Quinn, C. P. Pyrolysis of n-Butane. *Proc. R. Soc. London, Ser. A* **1962**, *270*, 267–284.
- (17) Sheng, C. Y.; Dean, A. M. Importance of Gas-Phase Kinetics within the Anode Channel of a Solid-Oxide Fuel Cell. *J. Phys. Chem. A* **2004**, *108*, 3772–3783.
- (18) Torok, J.; Sandler, S. Kinetics of Pyrolysis of n-Butane. *Can. J. Chem.* **1969**, *47*, 3863–3869.
- (19) Hughes, D. G.; Marshall, R. M.; Purnell, J. H. Rate Constants for Initiation of n-Butane Pyrolysis and for Recombination of Ethyl Radicals. *J. Chem. Soc., Faraday Trans.* **1974**, *70*, 594–599.
- (20) Mueller, J. E.; van Duin, A. C. T.; Goddard, W. A., III. Application of the ReaxFF Reactive Force Field to Reactive Dynamics of Hydrocarbon Chemisorption and Decomposition. *J. Phys. Chem. C* **2010**, *114*, 5675–5685.
- (21) Mallinson, R. G.; Braun, R. L.; Westbrook, C. K.; Burnham, A. K. Detailed Chemical Kinetics Study of the Role of Pressure in Butane Pyrolysis. *Ind. Eng. Chem. Res.* **1992**, *31*, 37–45.
- (22) van Duin, A. C. T.; Dasgupta, S.; Lorant, F.; Goddard, W. A. ReaxFF: A Reactive Force Field for Hydrocarbons. *J. Phys. Chem. A* **2001**, *105*, 9396–9409.
- (23) Tersoff, J. Empirical Interatomic Potential for Carbon, with Applications to Amorphous-Carbon. *Phys. Rev. Lett.* **1988**, *61*, 2879–2882.
- (24) Brenner, D. W. Empirical Potential for Hydrocarbons for Use in Simulating the Chemical Vapor-Deposition of Diamond Films. *Phys. Rev. B* **1990**, *42*, 9458–9471.
- (25) Mortier, W. J.; Ghosh, S. K.; Shankar, S. Electronegativity Equalization Method for the Calculation of Atomic Charges in Molecules. *J. Am. Chem. Soc.* **1986**, *108*, 4315–4320.
- (26) van Duin, A. C. T.; Strachan, A.; Stewman, S.; Zhang, Q.; Xu, X.; Goddard, W. A. ReaxFF(SiO) Reactive Force Field for Silicon and Silicon Oxide Systems. *J. Phys. Chem. A* **2003**, *107*, 3803–3811.