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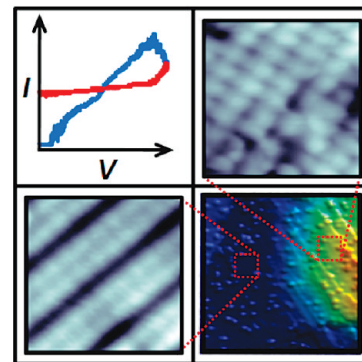
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ABSTRACT We have utilized tunneling electrons and thiophene adsorption to draw deuterium (D) from within the single-crystal bulk beneath Pd{110} up to subsurface adsorption sites. We found local faceting induced by this process, and determined the energy threshold of drawing bulk D to subsurface sites to be 0.38 ± 0.02 eV. We show that these facets propagate along the $\langle 1\bar{1}0 \rangle$ direction of the substrate, and that Pd{110} adopts the (1×1) surface reconstruction on the induced facets, yet maintains the paired row (1×2) structure on unaffected regions. After producing subsurface D, the facet plane tilts $3.2 \pm 0.8^\circ$ off the substrate plane.

SECTION Surfaces, Interfaces, Catalysis



Molecular-scale heterogeneous catalysis studies elucidate reaction mechanisms critical to technological processes, with the ultimate goal of improving catalyst design from the bottom up.¹ Major advances have resulted from the fabrication of metallic and bimetallic supported catalysts for a wide variety of catalytic reactions using shape-controllable synthetic approaches,^{2,3} and “self-activating” catalytic systems involving surface and subsurface hydrogen (H) or deuterium (D) in the selective partial hydrogenation of small alkenes.⁴ Transition and near-noble metals, specifically palladium (Pd), play important roles in catalyst design and function because of their behavior when modified by H or D adsorption and absorption.^{5–7} It is understood that both the physical and electronic landscape of the metal substrate influence the coincidence and interactions of adsorbed molecules through substrate-mediated interactions (SMIs), even at catalytically relevant temperatures.^{8–12} However, catalyst design is hindered by the complexity and incomplete understanding of the mechanisms of the catalytic reactions. For instance, hydrogenation on Pd is known to depend strongly on both the surface structure and on the distribution of heteroatom species on and beneath the metal surface.^{13–19} The dissolution of hydrogen into metal subsurface sites and its effects on surface adsorbates has been addressed in a review by Christmann and references therein highlighting ensemble spectroscopic techniques such as low-energy electron diffraction (LEED), Auger electron spectroscopy (AES), temperature-programmed desorption (TPD), and high-resolution electron energy loss spectroscopy (HREELS).²⁰ Additionally, the subsurface hydride of Pd{111} has been revealed directly by scanning tunneling microscopy (STM).^{16–18,21,22} A recent advance by Teschner et al. demonstrated that this subsurface species, along with subsurface carbon impurities, impact catalytic selectivity through localized surface structural transitions.¹⁹

In this work, we use the direct, local probe of our ultra-stable, extreme high vacuum (XHV) cryogenic (4 K) STM²³ to elucidate these surface transitions on the atomic scale and with direct local spectroscopic characterization. We study the D/Pd{110} catalytic system beginning with D-induced surface reconstruction and discover local faceting of the D/Pd{110}-(1 × 2) surface upon thiophene adsorption followed by application of tunneling electrons. Initial reconstruction of Pd{110} by D adsorption follows the (1×2) row pairing mechanism.²⁴ Subsequent adsorption of thiophene combined with the perturbation by the STM tip drives the formation of localized (1×1) facets tilting off the plane of the underlying substrate to facilitate the population of subsurface sites by D.

Induced facet reconstruction is known to occur on a variety of surfaces, both by molecular chemisorption^{25–29} and by epitaxial growth during metal deposition.^{30–32} Analyses by Madey and co-workers have followed surface reconstruction and faceting induced by epitaxial growth of ultrathin metallic films of Pd on tungsten and molybdenum surfaces.^{33–36} However, whether induced by molecular adsorption or by epitaxial growth, faceting in these instances occurs in a fashion parallel to the plane of the underlying substrate lattice. Out-of-plane faceting has been observed for vicinal Cu{100} in the presence of oxygen.³⁷ There are few examples of achieving local control via STM,^{38,39} and none address the impact of subsurface species on adsorbate-induced faceting. Thus, this work provides new insight into how adsorbates influence local structure of catalyst surfaces in the presence of surface and/or subsurface D, and how the

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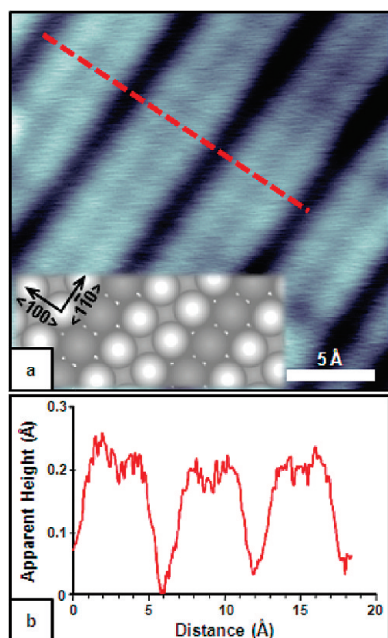


Figure 1. D-induced Pd{110}-(1 × 2) surface reconstruction. (a) Atomically resolved STM image (substrate bias voltage $V_s = 0.01$ V; tunneling current $I_t = 1.4$ nA; $22 \text{ Å} \times 22 \text{ Å}$) with model structure (inset) of the (1 × 2) phase. (b) Apparent height profile acquired along the dashed red line in panel a.

individual components of heterogeneous catalytic systems work cooperatively, ultimately to affect chemical reactions.

Previous studies of H (D) on Pd{110} have focused on understanding the array of induced in-plane substrate reconstructions upon chemisorption.^{40–49} Because Pd{110} is known to undergo surface reconstructions with and without H or D adsorption,^{49–51} it is not surprising that additional chemisorption or coadsorption of other molecules can drive surface atoms toward energetically more favorable arrangements. Further, uptake of H or D into subsurface sites causes 6% outward relaxation relative to the bulk Pd lattice constant.⁴⁸ This was observed via differential conductance (dI/dV) STM imaging of subsurface H in Pd{111}.¹⁸ In addition, a recent study by Kralj et al. followed, directly, the initial stages of H-induced reconstruction of Pd{110} in situ by STM at room temperature.⁵²

Here, we have chosen to study thiophene on D/Pd{110}-(1 × 2) because it provides insight into mechanisms of catalyst reactivity or poisoning, for instance in petroleum feedstocks, and because it has been used as a model toward elucidating the competing mechanisms of hydrogenation and hydrodesulfurization.^{53–56} We study this system at 4 K to observe surface transitions and intermediate states that might not be accessible even fleetingly or observable at higher temperatures.

Adding D to a clean Pd{110} substrate results in the typical (1 × 2) surface reconstruction (Figure 1a). Figure 1b shows the height profile taken along the dashed red line in panel a, highlighting the row-pairing of this surface reconstruction. We use this and other similar images to confirm the reconstruction, and to show that contamination by subsurface impurities is minimal with our preparation procedure, limited to small

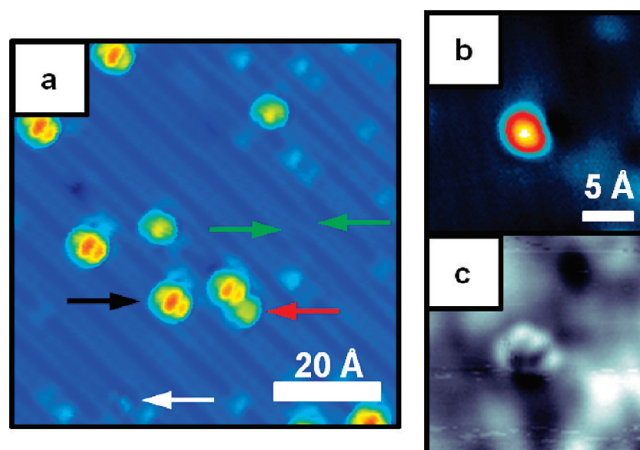


Figure 2. Thiophene adsorption on D/Pd{110}-(1 × 2) and on clean Pd{110}-(1 × 1). (a) Topographic STM image ($V_s = -0.015$ V; $I_t = 0.2$ nA; $73 \text{ Å} \times 73 \text{ Å}$) of thiophene deposited at low coverage on the D-reconstructed Pd{110}-(1 × 2) surface. Thiophene molecules appear as protrusions and adopt two distinct shapes/intensities in STM images depending upon adsorption site. (b) Topographic STM image ($V_s = 0.1$ V; $I_t = 0.9$ nA; $22 \text{ Å} \times 22 \text{ Å}$) of thiophene on Pd{110}-(1 × 1), accompanied by a simultaneously recorded differential conductance image, (c) showing enhancement of the local density of states (LDOS) at this energy. At positive sample bias, we observe lobes attributed to empty states.

amounts of atomic sulfur (S) contamination, as observed in our earlier work on Ni{110}²³ and in our previous investigations on Pd{111}.^{14,18} We observe a small degree of S contamination even prior to thiophene deposition. The difficulty in removing S impurities stems from its concentration in the bulk of the crystal and its tendency to migrate to the surface and subsurface regions during annealing. The high-resolution STM image shown in Figure 1 was acquired such that no S was observed in the region.

Once the (1 × 2) surface was confirmed, we deposited thiophene to address local structural and electronic perturbations. The STM image in Figure 2a shows surface morphology after 4 K deposition of thiophene. We observe thiophene adsorption atop the paired rows (black arrow, Figure 2a), with some molecules of lower apparent height within the troughs between paired rows (red arrow, Figure 2a). The (1 × 2) surface also exhibits “railroad switch” structural perturbations (green arrows in Figure 2a) that are a result of the reconstruction. The dI/dV image acquired over a single thiophene molecule on the (1 × 1) surface (Figure 2c) reveals electronic lobes, yielding strong evidence that this molecule is lying flat on the surface. Hence, molecules lying atop the paired rows see more (1 × 1) structure locally and adopt the flat configuration, whereas the molecules adsorbed between paired rows bond through the sulfur and thus adopt a tilted conformation. Similarly, two distinct bonding orientations were also observed in our previous study of thiophene on Ni{110}.⁵⁷ Surface-bound S impurities are also apparent, denoted by the white arrow. The adsorption distribution as a result of possible SMIs will be discussed elsewhere, but may be attributed to the competition between the surface physical structure (i.e., the (1 × 1) versus the (1 × 2) paired row reconstructions) and the surface electronic structure of the Pd substrate.

In order to observe both the molecular structure of the thiophene adsorbates and the atomic structure of the underlying Pd surface, we typically scan at relatively high tunneling currents (> 1 nA) and low sample biases (< 100 mV). On the clean (1×1) surface, we achieve the deconvolution of the molecular electronic structure (Figure 2c) acquired simultaneously with Figure 2b by dI/dV imaging. Because we image in constant-current mode and at positive sample bias, we observe empty electronic states. The observed lobes account for the π^* orbitals and are indicative of those observed in the STM simulations of thiophene on Pd{111}.⁵⁸

On the (1×2) reconstruction, at larger V_s , molecular detail is lost, but higher conductance features increasingly appear in the area over which the STM probe rests between images. In Figure 3, topographic STM images show the result of hovering the tip over a thiophene-covered (white protrusions) D-reconstructed Pd surface. The phenomenon observed here after hovering over the center of the scan frame for 2 min at 1 V is manifested in the diffuse, higher-contrast feature in the center of the frame on the right (dotted circle in the right image of Figure 3). Subsequent hovering and scanning produces more prominent perturbations. Since other subsurface impurities

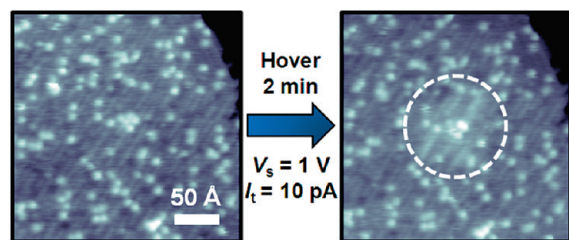


Figure 3. Drawing D to the subsurface sites of Pd{110}. The STM topographic image ($V_s = 0.5$ V; $I_t = 0.010$ nA; $260 \text{ Å} \times 260 \text{ Å}$) on the left was acquired before setting the bias voltage to 1 V. After setting the bias voltage to 1 V and having the STM tip hover in its rest position over the center of the frame for 2 min, the image on the right was acquired. A diffuse protruding region (dashed white circle) was observed at the center of the image.

are minimized by careful preparation, we attribute this effect to the population of subsurface sites by D atoms drawn up from the bulk Pd. Previously, we produced and imaged the subsurface hydride of Pd{111} at 4 K. We have shown that we can use ballistic scattering of tunneling electrons to draw and to manipulate bulk hydrogen to the stable subsurface region.¹⁸ Whereas subsurface H in Pd{111} causes outward relaxation of the surface lattice on the order of a few hundredths of an Ångström, here, we observe a much different effect—one that occurs along the direction of the paired rows (the $\langle 1\bar{1}0 \rangle$ direction) only in the presence of additional thiophene. In the absence of preadsorbed D, deposition of thiophene and subsequent imaging resulted in no apparent local structural perturbation (see Supporting Information, Figure S1a). Similarly, imaging the D-reconstructed surface shown in Figure 1 at larger V_s (3 eV) in the absence of thiophene also produced no faceting (see Supporting Information, Figure S1b). It should be noted that the image in Figure S1b was acquired after obtaining 40 individual conductance spectra along the trajectory indicated by the red arrow from +1 to +4 eV with a total acquisition time of over 4 h. This energy range is well above the faceting threshold. Thus, we posit that adsorption of thiophene onto the D-reconstructed surface lowers the overall barrier to near-surface D diffusion, and may be coverage dependent, as indicated by the work in ref 4.

Significant restructuring of the surface occurs with repeated imaging, and is both energetically and kinetically driven. That is, we observe faster growth at higher V_s ; restructuring also occurs at lower V_s over longer times, but not below a threshold bias voltage (*vide infra*). New faceted steps propagate along the $\langle 1\bar{1}0 \rangle$ direction with angularity incommensurate with the plane of the underlying substrate. Step edges are typically more catalytically active than terrace sites due to the charge dipole across the step and higher available coordination.^{59–72} Here, the formation of new steps may be driven by the favorability of thiophene to be in proximity to these active sites in the presence of subsurface D. Figure 4 shows one such example of this surface restructuring,

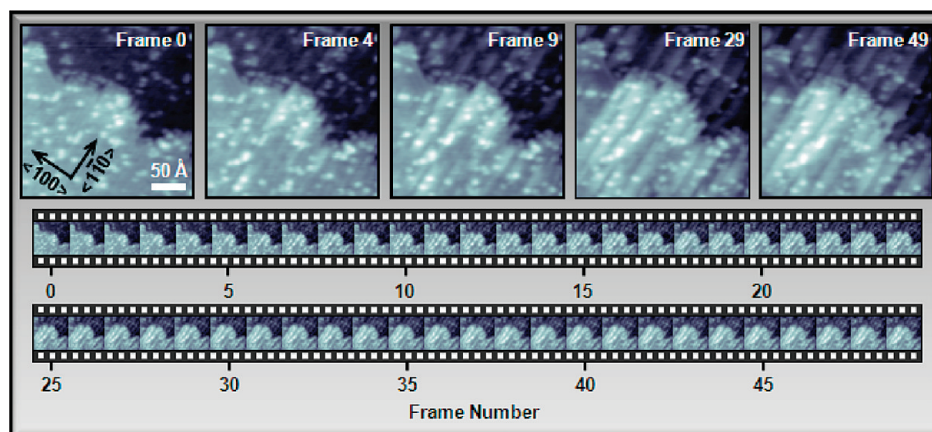


Figure 4. Selected frames extracted from a series of sequential STM images ($V_s = 1.0$ V; $I_t = 0.05$ nA; $260 \text{ Å} \times 260 \text{ Å}$) that were rendered into a time-lapse video file (Supporting Information). Already evident at the beginning of the sequence (Frame 0) is a more protruding region at the center of the image, where the STM probe rests for a short time (< 10 ms) before and between frame acquisition. Subsurface sites are increasingly populated as the tip rasters across the surface at $V_s = 1$ V, and facets emerge and propagate along the $\langle 1\bar{1}0 \rangle$ direction of the Pd{110} substrate. Population of the subsurface sites by D was saturated by frame 30.

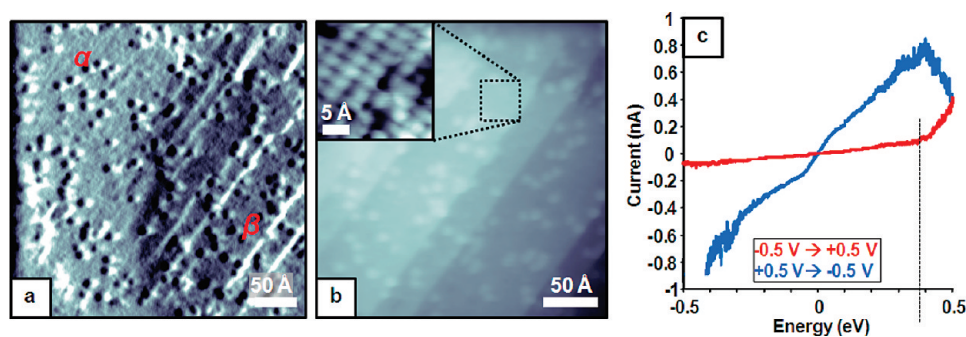


Figure 5. Determining the onset of the population of subsurface D in the presence of thiophene. (a) STM topographic (derivative) image ($V_s = 0.05 \text{ V}$; $I_t = 0.05 \text{ nA}$; $348 \text{ Å} \times 348 \text{ Å}$) highlighting the deviation of the facet planes (β) from the underlying substrate plane (α). (b) Topographic STM image ($V_s = 0.05 \text{ V}$; $I_t = 0.05 \text{ nA}$; $260 \text{ Å} \times 260 \text{ Å}$) of region β in panel a, plane subtracted (flattened by tilting the image). The inset is a high-resolution STM topographic image ($V_s = 0.01 \text{ V}$; $I_t = 1.4 \text{ nA}$; $22 \text{ Å} \times 22 \text{ Å}$) acquired over the region denoted by the black dotted box. Here, the $\text{Pd}\{110\}$ -(1×1) structure has returned. The protrusions in both panels a and b are thiophene molecules. (c) Representative $I(V)$ spectrum taken over Pd in area α of panel a ($V_{\text{gap}} = 0.05 \text{ V}$; $I_{\text{gap}} = 0.05 \text{ nA}$). The red curve represents the forward bias sweep direction from negative to positive bias, and the blue curve represents the reverse sweep. Further spectra acquired at the same spot retraced the blue curve (see Figure S2 in the Supporting Information). The dashed line marks the onset value ($0.38 \pm 0.02 \text{ V}$) acquired from an average of 45 forward spectra at fresh locations on the surface.

highlighting individual images from 50 consecutive images acquired at 1 V. Faceting is nearly complete by frame 29, and little molecular diffusion is observed during faceting. Several time-lapse sequences acquired over up to 7 h and at a variety of sample bias voltages demonstrate that faceting is reliably induced at $|V_s| > 500 \text{ mV}$.

To demonstrate that faceting is localized to the region imaged by the STM tip, we first created the facets at 1 V and then repositioned the tip to image both flat and faceted areas at small V_s . Scanning at lower V_s prevents further subsurface D population. Figure 5a shows an area comprising both the (1×2) paired row surface (denoted α) and the faceted surface induced by scanning at 1 V (denoted β). These areas are shown prior to investigation by local $I(V)$ tunneling spectroscopy. Figure 5a is displayed in derivative mode to highlight the angularity of the stepped facets with respect to the flat underlying substrate. Using trigonometry, height profiles acquired and averaged over topographic images in this area (one such profile acquired over the dotted red line in Figure 5a) reveal that the faceted plane is tilted $3.2 \pm 0.8^\circ$ off the substrate plane. This small value is greatly exaggerated by the display used, as observed in Figure 5a, and accounts for the (1×1) structure over the β region.

Figure 5b is an image acquired over the β region, plane subtracted (flattened by tilting the image) to demonstrate that the angles of the faceted planes are identical within experimental error. The inset of Figure 5b shows a high-resolution scan resolving the Pd atoms in the (1×1) configuration over a faceted step, further confirmed by a fast Fourier transform (FFT) of the inset of Figure 5b (see Figure S3 in the Supporting Information).

Scanning tunneling spectroscopy (STS) over the paired row $\text{Pd}\{110\}$ enables us to determine the onset voltage for D migration to the subsurface layer. The spectra in Figure 5c represent typical $I(V)$ curves for the faceting behavior (forward sweep in red and reverse sweep in blue) acquired over the α region. Bulk D is drawn to the subsurface region on the initial forward sweep, causing outward relaxation of the Pd lattice, thus, physically decreasing the probe-sample separation. Because of the exponential dependence between current and

probe-sample separation, and because the STM probe is held at a fixed position during spectrum acquisition, this surface relaxation is observed as a steep onset in current response (dashed vertical line in Figure 5c). Since only a finite amount of bulk D can be drawn up into the subsurface region beneath the probe, we are able to acquire only one such spectrum in each location. Each spectrum displays the indicative onset in the forward sweep direction and the reconstructed surface signature on the reverse sweep. From 45 such spectra, we have determined the onset of drawing subsurface D to be $0.38 \pm 0.02 \text{ eV}$. This $I(V)$ behavior is not observed over the (1×1) faceted regions of β , suggesting complete population of the subsurface sites with D. Subsequently recorded spectra have overlapping forward and reverse sweeps, as shown in Figure S2 in the Supporting Information. We point out a plausible correlation here with the STM images observed by Katano et al. (specifically that of Figure 8c of ref 4.). Their annealing at 202 K after additional molecular adsorption onto hydrogen precovered $\text{Pd}\{110\}$ resulted in a vicinal surface. Here, we believe that we are observing the beginnings of this surface transformation by injecting tunneling electrons of sufficient energy relative to the thermal energy of annealing.

In this work, we have shown the propensity of the $\text{D}/\text{Pd}\{110\}$ -(1×2) to undergo faceting along the $\langle 1\bar{1}0 \rangle$ direction after depositing thiophene and during observation by STM resulting from subsurface site population by D. At 4 K, this process requires additional local molecular adsorption in order to reduce the total surface free energy that might otherwise be achieved thermally, as in ref 4. Spectroscopic analysis reveals the onset of drawing subsurface D to be $0.38 \pm 0.02 \text{ eV}$. Regions that do not show appreciable subsurface D exhibit the expected (1×2) surface reconstruction upon D adsorption, whereas faceted areas adopt the (1×1) atomic Pd configuration $3.2 \pm 0.8^\circ$ tilted with respect to the substrate plane. These differences in local surface structure account for the preferred adsorption sites found for thiophene between rows. Differential conductance imaging has elucidated the electronic LDOS of thiophene adsorbates, and may provide further insight into the bonding mechanism of this

and analogous heterocycles onto transition metal surfaces during hydrogenation, desulfurization, and deoxygenation.

We continue our studies of this system in order to develop a more comprehensive understanding of the catalytic reactions of small molecules on transition metals at the single-molecule scale, extending our spectroscopic capabilities to enhanced single-molecule vibrational measurements, theoretically modeled approaches, and subsequent action spectra.^{10,14,73} This phenomenon has implications in elucidating key aspects of heterogeneous catalytic processes, including substrate structure changes,^{74,75} directed modification of surface potential energy landscapes, and bulk diffusion. Additionally, these studies may provide insight into the mechanism responsible for metal embrittlement, and for developing routes toward stable, efficient, and clean hydrogen-storage materials.

EXPERIMENTAL DETAILS

All experiments were carried out in a custom-built XHV (base pressure < 10⁻¹² Torr), ultrastable, low temperature STM system with the additional capability of sample rotation for direct deposition of gas-phase molecules from the room temperature chamber onto the sample held at 4 K.²³ Tungsten probes were mechanically cut and sharpened in situ by high-voltage pulses.⁷⁶ The Pd{110} single crystal (MaTeck, GmbH) was prepared by cycles of Ar⁺ sputtering at 875 K, oxygen treatment (~2 × 10⁻⁶ Torr at 875 K), and flash annealing to 1175 K. Sample cleanliness was ascertained by STM imaging prior to D or thiophene deposition. Molecular deuterium (Matheson Tri Gas, Parsippany, NJ) was deposited directly from a room temperature chamber onto the sample held at 4 K using a high-precision sapphire leak valve. Neat thiophene (Supelco, Bellefonte, PA) was further purified by several freeze–pump–thaw cycles and deposited directly from an identical high-precision sapphire leak valve. Purity and deposition of both deuterium and thiophene were monitored via a quadrupole residual gas analyzer (QMA 125, Balzers, Liechtenstein). The standard deposition parameters (deuterium, 180 Langmuir; thiophene, 480 Langmuir) were calibrated using an inverted magnetron cold cathode ion gauge at the main chamber, although the actual deposition incident on the sample can be up to 5 orders of magnitude lower due to the efficiency of cryopumping.²³ Ion pumps and gauges were turned off during actual deposition to avoid molecular fragmentation, and partial pressure was maintained by differentially pumping via a turbomolecular pump. Differential conductance imaging was performed by superimposing a small AC modulation ($\nu_{AC} = 1$ kHz, $V_{rms} = 25$ mV) on the DC sample bias, and then by monitoring the demodulated signal using a lock-in amplifier (SR850, Stanford Research Systems, Sunnyvale, CA). Topography and differential conductance images were acquired simultaneously. All bias voltages reported are sample biases (V_s) unless stated otherwise.

SUPPORTING INFORMATION AVAILABLE STM images of thiophene on Pd{110} in the absence of D and of the D-reconstructed surface in the absence of thiophene (Figure S1a,b, respectively), conductance spectra acquired over a (1 × 1) faceted area (Figure S2), the FFT of the STM image acquired over this area (Figure S3), and the video file showing increasing population of subsurface sites by D

in Figure 4. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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