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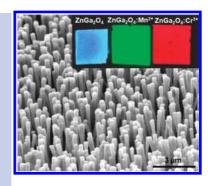


## Red, Green, and Blue Luminescence from ZnGa<sub>2</sub>O<sub>4</sub> Nanowire Arrays

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**ABSTRACT** Large-area (>1 cm²), vertically aligned ZnGa<sub>2</sub>O<sub>4</sub>, Mn²+-doped ZnGa<sub>2</sub>O<sub>4</sub> (ZnGa<sub>2</sub>O<sub>4</sub>:Mn²+) and Cr³+-doped ZnGa<sub>2</sub>O<sub>4</sub> (ZnGa<sub>2</sub>O<sub>4</sub>:Cr³+) nanowire arrays were synthesized by a two-step thermal evaporation method using ZnO nanowire arrays as the templates followed by reaction with Ga and/or dopant (Mn or Cr) vapors. Bright red, green, and blue luminescence was achieved, respectively, from ZnGa<sub>2</sub>O<sub>4</sub>:Cr³+, ZnGa<sub>2</sub>O<sub>4</sub>:Mn²+, and ZnGa<sub>2</sub>O<sub>4</sub> nanowire arrays under ultraviolet light irradiation. The realization of three primary colors from one host material suggests that full color display based on ZnGa<sub>2</sub>O<sub>4</sub> nanowires might be achievable.



**SECTION** Nanoparticles and Nanostructures

uture flat panel display technologies demand highly efficient and stable phosphor materials for better performance. <sup>1</sup> Zinc gallate (ZnGa<sub>2</sub>O<sub>4</sub>), a ternary spinel material with a band gap of 4.4-4.7 eV, has shown excellent potential for incorporation in future display system because of its prominent blue emission, high chemical and thermal stability, and good cathodoluminescence characteristics at low-voltage. 2,3 ZnGa2O4 is naturally a very promising blueemitting phosphor.4 It can also act as an excellent host material for high luminance multicolor-emitting phosphors when doped with appropriate activators: Mn-activated  $ZnGa_2O_4$  ( $ZnGa_2O_4$ : $Mn^{2+}$ ) for green emission<sup>5,6</sup> and Crdoped ZnGa<sub>2</sub>O<sub>4</sub> (ZnGa<sub>2</sub>O<sub>4</sub>:Cr<sup>3+</sup>) for red emission.<sup>5,7</sup> ZnGa<sub>2</sub>O<sub>4</sub> phosphors (with or without doping) were usually used in the form of thin films.<sup>8-11</sup> It is anticipated that when the phosphors are made into nanocrystals, the density of emission sites would be dramatically increased, which may lead to high-brightness and high-definition display. 12

Considerable efforts have been spent in recent years to fabricate  $\rm ZnGa_2O_4$  nanowires and their arrays.  $^{13-15}$  Bae et al. synthesized aligned  $\rm ZnGa_2O_4$  nanowires by thermal evaporation of  $\rm ZnO-Ga$  powders using Au as a catalyst.  $^{13}$  Chang and Wu reported a three-step process using  $\rm Ga_2O_5$  nanowires as the templates.  $^{14}$  All the so-far demonstrated  $\rm ZnGa_2O_4$  nanowires are undoped, so the observed luminescence is the self-activated blue emission stemming from the octahedral  $\rm Ga-O$  groups in the spinel lattices.  $^{4.16}$  For applications in full color displays and nanoscale light sources, however, doping appropriate activators (e.g.,  $\rm Mn^{2+}$  and  $\rm Cr^{3+})$  into  $\rm ZnGa_2O_4$  nanowires is highly desirable.

Recently, ZnO nanowires were frequently used as templates to fabricate a series of ZnO-based ternary spinel

nanotubes (e.g., ZnAl<sub>2</sub>O<sub>4</sub>) or core/shell nanowires (e.g., ZnO/Zn<sub>2</sub>TiO<sub>4</sub>) using a three-step process, <sup>16-18</sup> in which presynthesized ZnO nanowires were coated with an oxide (e.g., Al<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub>) shell by various deposition techniques (e.g., atomic layer deposition) followed by solid-state reaction of ZnO with oxide shell at high temperature. Depending on the oxide shell thickness and diffusion direction (outward or inward relative to the ZnO nanowire center), the resulting product is either in the tube form or in a core/shell structure. In this three-step process, an oxide shell deposition stage is needed, and the doping of the ternary nanotubes or core/ shell nanowires has not yet been achieved. In this letter, we report the fabrication of highly aligned ZnGa<sub>2</sub>O<sub>4</sub> nanowire arrays and the first doping of ZnGa<sub>2</sub>O<sub>4</sub> nanowires with transition metals Mn and Cr using presynthesized ZnO nanowire arrays as the templates via a simpler two-step thermal evaporation method. The three primary colors (red, green, and blue: RGB) that are required for full color displays were realized from these doped and undoped ZnGa<sub>2</sub>O<sub>4</sub> nanowire arrays: red from ZnGa<sub>2</sub>O<sub>4</sub>:Cr<sup>3+</sup>, green from ZnGa<sub>2</sub>O<sub>4</sub>:Mn<sup>2+</sup>, and blue from undoped ZnGa<sub>2</sub>O<sub>4</sub>.

Figure 1a,b shows the scanning electron microscope (SEM) images of the aligned ZnO nanowire arrays and the ZnGa $_2$ O $_4$  nanowire arrays fabricated at 1000 °C, respectively. It is clear that the ZnGa $_2$ O $_4$  nanowires reserve the alignment and sizes of the ZnO nanowires very well. Depending on the ZnO nanowire templates, the obtained ZnGa $_2$ O $_4$  nanowires have diameters in the range of 50–200 nm and lengths up to

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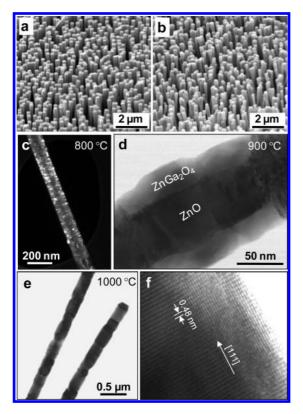
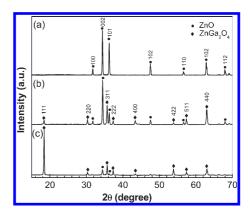


Figure 1. (a) SEM image of a ZnO nanowire array. (b) SEM image of a ZnGa $_2$ O $_4$  nanowire array fabricated at 1000 °C. (c) Z-contrast TEM image of a ZnO/ZnGa $_2$ O $_4$  core/shell nanowire obtained at 800 °C. The bright particles are ZnGa $_2$ O $_4$ . (d) TE mode of a ZnO/ZnGa $_2$ O $_4$  core/shell nanowire formed at 900 °C. (e) Low-magnification TEM image of two ZnGa $_2$ O $_4$  nanowires fabricated at 1000 °C. (f) HRTEM image of a ZnGa $_2$ O $_4$  nanowire.

 $\sim$ 5  $\mu$ m. Transmission electron microscope (TEM) analyses of the samples formed at different temperatures (800°, 900°, and 1000 °C) show that the conversion reaction of ZnO +  $Ga + O_2 \rightarrow ZnGa_2O_4$  is determined by the reaction temperature. At 800 °C, only a thin ZnGa<sub>2</sub>O<sub>4</sub> particle layer was formed on the ZnO nanowire surface (Figure 1c). At 900 °C, about half of ZnO was converted to ZnGa $_2$ O $_4$  (Figure 1d). At 1000 °C, the ZnO nanowires were completely converted to ZnGa<sub>2</sub>O<sub>4</sub> nanowires (Figure 1e). Different from the starting ZnO nanowires which is a single crystal, 19 the ZnGa<sub>2</sub>O<sub>4</sub> nanowires are made of many  $0.1-1 \mu m$  long crystallites that stack one-byone along the nanowires' longitudinal direction (Figure 1e). Unlike the previously reported ZnO-based ternary nanowires and nanotubes prepared by the three-step process, 16-18 no voids or tubular structures were observed in our ZnGa<sub>2</sub>O<sub>4</sub> nanowires formed at 1000 °C. This means that the Ga-ZnO reaction at 1000 °C is very efficient and sufficient, and that the inward diffusion of Ga ions to the ZnO nanowire center is the dominant diffusion process. 16 High-resolution TEM (HRTEM) analysis (Figure 1f) shows that each crystallite is a ZnGa<sub>2</sub>O<sub>4</sub> single crystal with a growth direction along (111). The surfaces of the ZnGa<sub>2</sub>O<sub>4</sub> nanowires are clean, atomically sharp, and without any sheathed secondary phases.

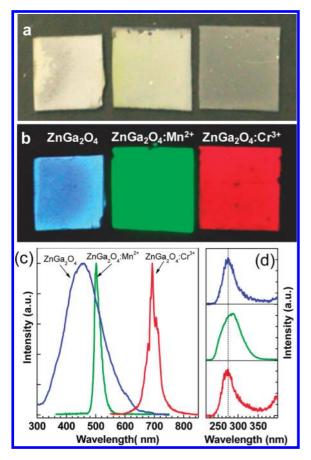
The temperature-dependent ZnO to  $ZnGa_2O_4$  conversion was verified by X-ray diffraction (XRD) analyses, as that



**Figure 2.** XRD patterns of (a) ZnO nanowire arrays, (b) a ZnO/ZnGa $_2$ O $_4$  core/shell nanowire array prepared at 800 °C, and (c) a ZnGa $_2$ O $_4$  nanowire array formed at 1000 °C.

shown in Figure 2. At 800 °C, only a small amount of  $ZnGa_2O_4$  phases was formed in the product (Figure 2b), while, after 30 min of reaction at 1000 °C, the  $\langle 001 \rangle$ -oriented wurtzite ZnO nanowires (Figure 2a) were totally converted to spinel  $ZnGa_2O_4$  nanowires with a preferential growth direction along  $\langle 111 \rangle$  (Figure 2c). The remaining small amount of ZnO phases in Figure 2c is believed to come from the ZnO layer located at the bottom of the nanowire array,  $^{19}$  where the penetration of ZnO Ga vapor is limited and thus the conversion reaction is incomplete. This temperature-dependent phase conversion is in good agreement with the results obtained by annealing ZnO and ZnO powders in air at ZnO 1000 °C.

Intentional introduction of impurities (activators) into inorganic phosphors is fundamental to control their optical properties. <sup>21</sup> This has worked very well for ZnGa<sub>2</sub>O<sub>4</sub> powders: undoped ZnGa<sub>2</sub>O<sub>4</sub> powder is a blue-emitting phosphor, while the Mn<sup>2+</sup>- and Cr<sup>3+</sup>-activated ZnGa<sub>2</sub>O<sub>4</sub> powders emit bright green<sup>5,6</sup> and red<sup>5,7</sup> luminescence, respectively. Our present experiments further show that this doping principle is also true for ZnGa<sub>2</sub>O<sub>4</sub> nanowires. Adding a small amount of MnCl<sub>2</sub> or CrCl<sub>3</sub> powder into the source does not affect the morphology, alignment, and crystal structures of the ZnGa<sub>2</sub>O<sub>4</sub> nanowires. However, the optical properties of the ZnGa<sub>2</sub>O<sub>4</sub> nanowires are hence dominated by the optically active Mn<sup>2+</sup> or Cr<sup>3+</sup> ions. Figure 3a is a digital image of the undoped, Mn-doped, and Cr-doped ZnGa<sub>2</sub>O<sub>4</sub> nanowire arrays grown on Si substrates (size:  $\sim$ 1 cm  $\times$  1 cm). Under the excitation of a 254 nm ultraviolet (UV) lamp, bright blue, green, and red luminescence (Figure 3b) emits from the undoped ZnGa<sub>2</sub>O<sub>4</sub>, ZnGa<sub>2</sub>O<sub>4</sub>:Mn<sup>2+</sup>, and ZnGa<sub>2</sub>O<sub>4</sub>:Cr<sup>3+</sup> nanowire arrays, respectively. The uniform luminescence brightness across the 1 cm<sup>2</sup> area suggests that the doping is very sufficient and uniform. Figure 3c is the room-temperature photoluminescence (PL) spectra of the three ZnGa<sub>2</sub>O<sub>4</sub> nanowire arrays under 275 nm UV light excitation. The blue emission from the undoped ZnGa<sub>2</sub>O<sub>4</sub> nanowires is broad and peaked at 452 nm, which can be attributed to the self-activation center of the octahedral Ga-O group in the spinel lattices. 4,16 The green emission band from the ZnGa<sub>2</sub>O<sub>4</sub>:Mn<sup>2+</sup> nanowire array is narrow and centered at 503 nm, which can be clearly assigned to the  $\mathrm{Mn}^{2+4}\mathrm{T_1}-{}^6\mathrm{A_1}$  characteristic transition. <sup>5,6</sup> The red emission from the ZnGa<sub>2</sub>O<sub>4</sub>:Cr<sup>3+</sup> nanowire array is peaked at 696 nm



**Figure 3.** (a) Digital image of (from left to right) undoped  $\rm ZnGa_2O_4$ ,  $\rm ZnGa_2O_4$ :Mn²+, and  $\rm ZnGa_2O_4$ :Cr³+ nanowire arrays grown on Si substrates (size: 1 cm  $\times$  1 cm). (b) Optical image of the samples in panel a under a 254 nm UV lamp irradiation. (c) PL spectra of the three  $\rm ZnGa_2O_4$  nanowire samples. The excitation wavelength is 275 nm. (d) Excitation spectra of (from top to bottom)  $\rm ZnGa_2O_4$ ,  $\rm ZnGa_2O_4$ :Mn²+, and  $\rm ZnGa_2O_4$ :Cr³+ nanowire arrays.

and can be assigned to the  $^2E-^4A_2$  characteristic transition of  $Cr^{3+}$  ions.  $^{5.7}$  Meanwhile, when monitoring the emission maximum of the three  $ZnGa_2O_4$  products, the characteristic excitation peak of the  $ZnGa_2O_4$  host centered at 275 nm emerges in the respective excitation spectrum (Figure 3d), indicating that  $Mn^{2+}$  and  $Cr^{3+}$  ions have been effectively doped into the  $ZnGa_2O_4$  nanowires and respectively substitute for the tetrahedral site of  $Zn^{2+}$  and the octahedral site of  $Ga^{3+}$   $^{22,23}$ 

To determine the concentration and confirm the oxidation states of Mn and Cr ions in  $\rm ZnGa_2O_4$  nanowires, we conducted analyses using X-ray photoelectron spectroscopy (XPS). The XPS composition measurements show that the Mn or Cr concentration in the doped  $\rm ZnGa_2O_4$  nanowires is about 0.3–0.5 atom %, a reasonable level for an efficient phosphor. Figure 4a and 4b show the XPS spectra of Mn ions and Cr ions in doped  $\rm ZnGa_2O_4$  nanowires, respectively. For the Mn XPS spectrum shown in Figure 4a, the peaks at 653.4 and 641.4 eV can be attributed to Mn  $\rm 2p_{1/2}$  and Mn  $\rm 2p_{3/2}$  states, respectively. These binding energies are in good agreement with the reported values of Mn  $\rm ^{2+}$  ions,  $\rm ^{24}$  indicating that the

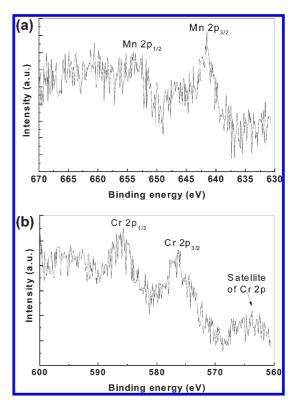


Figure 4. XPS spectra of (a)  $\rm Mn^{2+}$  ions and (b)  $\rm Cr^{3+}$  ions in  $\rm ZnGa_2O_4$  nanowires.

Mn ions in our ZnGa $_2$ O $_4$  nanowires are in divalent (Mn $^{2+}$ ) oxidation state. For the Cr XPS spectrum shown in Figure 4b, the peaks at 585.8 and 576.3 eV correspond to Cr  $2p_{1/2}$  and Cr  $2p_{3/2}$  states, respectively. In addition to these two peaks, a low binding energy satellite peak of Cr 2p located at 563 eV was also observed. These peaks indicate that the Cr ions in the ZnGa $_2$ O $_4$  nanowires are in the trivalent (Cr $^{3+}$ ) oxidation state.

In summary, using a simple two-step reaction process, we have demonstrated the synthesis of large-area ZnGa<sub>2</sub>O<sub>4</sub>, ZnGa<sub>2</sub>O<sub>4</sub>:Mn<sup>2+</sup>, and ZnGa<sub>2</sub>O<sub>4</sub>:Cr<sup>3+</sup> nanowire arrays, which respectively emit bright blue, green, and red luminescence under UV light irradiation. The realization of three primary colors from one host material suggests that full color display based on ZnGa<sub>2</sub>O<sub>4</sub> nanowires might be achievable. These luminescent nanowires could also be used in applications such as solid-state lighting as well as radiation detection. The cathodoluminescence properties of the ZnGa<sub>2</sub>O<sub>4</sub> nanowires will be studied in future research. Moreover, the two-step methodology introduced in this letter may be applied to synthesize many other ZnO-based spinel oxides. For instance, our recent synthesis results show that undoped and Mn<sup>2+</sup>doped Zn<sub>2</sub>GeO<sub>4</sub> nanowires can be readily synthesized using the two-step method.

### **Experimental Method**

The two-step material synthesis process was conducted inside a well-controlled tube furnace system. Vertically aligned ZnO nanowires were first grown on a Si substrate (size:  $\sim$ 1 cm  $\times$  1 cm) by a low-temperature (600 °C) oxidation



method. 19 ZnGa<sub>2</sub>O<sub>4</sub> nanowire arrays were then fabricated through the reaction between ZnO nanowires and Ga vapor, where the pregrown ZnO nanowire arrays acted as the template, molten Ga (~1 g) was used as the Ga source, and the tiny amount of residue oxygen in the reaction chamber acted as the O source. The reaction was conducted at 800-1000 °C under an argon flow rate of 200 mL/min and a reaction chamber pressure of ~200 Torr for 30 min. For Mn or Cr doping, a tiny amount (~0.01 g) of MnCl<sub>2</sub> or CrCl<sub>3</sub> powders was added to the source. The products were analyzed by SEM (FEI Inspect F FEG-SEM), TEM (Hitachi HF-3300 FEG STEM at 300 kV), XRD (PANalytical X'Pert PRO diffractometer with Cu  $K\alpha$  radiation), spectrofluorometry (Horiba FluoroLog3-2iHR320), and XPS [ESCALab220i-XL spectrometer with a twin-anode Al K $\alpha$  (1486.6 eV) X-ray source]. For XPS measurement, all spectra were calibrated to the binding energy of the adventitious C1s peak at 284.6 eV.

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