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Gd(III) doping effect on magnetization and water proton relaxivities in ultra small iron oxide nanoparticles

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Two samples of ultra small Gd(III) doped iron oxide nanoparticles were prepared to investigate Gd(III) doping effect on longitudinal (r_1) and transverse (r_2) water proton relaxivities. Gd(III) doping mole percents were 0.2 and 0.4 for samples 1 and 2, respectively. Average particle diameters were 2.5 to 2.1 nm for samples 1 and 2, respectively. Reduced r_1 and r_2 values were observed in both samples. We attributed this to reduced magnetizations arising from opposing effect of Gd(III) to net magnetizations of Fe(III)/Fe(II) in oxide nanoparticles. © 2013 Author(s). All article content, except where otherwise noted, is licensed under a Creative Commons Attribution 3.0 Unported License. [<http://dx.doi.org/10.1063/1.4813307>]

I. INTRODUCTION

Magnetic properties of nanoparticles can depend on particle diameter because of anomalous surface spin effects related to unsaturated coordination of surface elements, surface oxidation, and surface coating.¹⁻⁴ Magnetic properties may also change with doping.

This work deals with Gd(III) doping in ultra small iron oxide nanoparticles. Note that Gd(III) has the largest 7 unpaired 4f electrons in the periodic table and thus, Gd(III) is regarded as the best element for T₁ MRI contrast agents because it strongly enhances the longitudinal water proton relaxation.^{5,6} Furthermore, gadolinium oxide nanoparticles have shown r_1 values which are much larger than those of Gd(III)-chelates due to a dense population of Gd(III) ions in a nanoparticle.^{7,8} Therefore, it is expected that Gd(III) doping might increase a longitudinal water proton relaxivity (r_1) in ultra small iron oxide nanoparticles (high spin doping effect). On the other hand, Gd(III) is known to oppose net magnetic moment of Fe(III)/Fe(II) in oxides, reducing magnetizations (reduced magnetization effect).⁹⁻¹³ Such an opposing effect had been also observed in Fe thin films covered with Gd.¹⁴⁻¹⁶ Therefore, these two effects will affect longitudinal (r_1) and transverse (r_2) water proton relaxivities, which was explored in this work.

In the previous work we developed ultra small iron oxide nanoparticles with an average particle diameter of 1.7 nm by refluxing Fe (III) precursor in triethylene glycol under air flow.¹⁷ In this work we additionally doped tiny mole percents of Gd(III) into ultra small iron oxide nanoparticles. Core particle diameter, crystal structure, surface composition, surface coating, magnetic properties, r_1 and

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r_2 values, and map images were measured by using various techniques. Finally, Gd(III) doping effect on r_1 and r_2 values were investigated.

II. EXPERIMENTAL

A. Preparation

Two samples of ultra small Gd(III) doped iron oxide nanoparticles were prepared. In order to prepare sample 1, 0.1 mmol of $\text{GdCl}_3 \cdot x\text{H}_2\text{O}$, 10 mmol of $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$, and 5 mmol of PEG diacid ($M_n = 600$) were added to 40 ml of triethylene glycol at room temperature. Reaction solution was magnetically stirred at 250°C for 24 hours under air bubbling. In order to prepare sample 2, 1.0 mmol of $\text{GdCl}_3 \cdot x\text{H}_2\text{O}$ was used. The other reaction conditions were the same as the sample 1. After cooled to room temperature, product solutions were washed with 400 ml of distilled water twice. To do this, 400 ml of distilled water was added to the reaction solutions and then centrifuged. Top solutions were decanted and the remaining solutions were diluted with distilled water again for another washing. Halves of sample solutions were re-dispersed into distilled water for water proton relaxivity and map image measurements and the remaining samples were dried in air to obtain powder samples for various characterizations.

B. Characterization

Exact mole percents of Gd and Fe in nanoparticles were determined by using an inductively coupled plasma atomic emission spectrometer (ICPAES) (Thermo Jarrell Ash Co., IRIS/AP). In order to measure these, ~ 1 ml of each sample solution was pre-treated with 4 ml of HCl (or 3 ml of HCl and 1 ml of HClO_4) and then the solution was kept to $50\text{--}60^\circ\text{C}$ until nanoparticles were completely dissolved in solution. 2–3% HNO_3 solution was added to the above solution to dilute it. The final solution was weighed and then used for ICPAES measurement. The core particle diameters of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles were measured by using a high resolution transmission electron microscope (HRTEM) (JEOL, JEM 2100F, 200 kV acceleration voltage). Nanoparticles dispersed in distilled water were dropped onto a 200-mesh copper grid covered with an amorphous carbon membrane by using a micropipette (Eppendorf, 2–20 μL), which was dried in air before loading into the microscope. A dynamic light scattering (DLS) particle size analyzer (UPA-150, Microtrac) was used to measure the hydrodynamic diameters of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles dispersed in triply distilled water. The sample solution concentration used to measure hydrodynamic diameters was ~ 0.1 mM Fe. The crystal structure of powder samples was measured by using an X-ray diffraction (XRD) spectrometer (Philips, X-PERT PRO MRD). XRD patterns were recorded by using an unfiltered $\text{CuK}\alpha$ ($\lambda = 0.154184$ nm) radiation for $2\theta = 15\text{--}90^\circ$. The surface coating of PEG diacid on nanoparticle surfaces was investigated by using a Fourier transform infrared (FT-IR) absorption spectrometer (Perkin Elmer, Spectrum GX & Auto Image) and a thermogravimetric analyzer (TGA) (TA Instruments, SDT-Q600). In order to record FT-IR absorption spectra, sample discs were prepared by pressing mixtures of KBr and powder samples. FT-IR absorption spectra were scanned between 400 and 4000 cm^{-1} . TGA curves were recorded to estimate the surface coating amounts and scanned between room temperature and 700°C under air flow. The surface composition of Gd and Fe in nanoparticles was measured by using an X-ray photoelectron spectrometer (XPS) (ULVAC-PHI, Quantera SXM) with an $\text{Al K}\alpha$ X-ray source ($=1486.6$ eV). Dried powder samples were loaded onto a carbon tape (5×5 mm) and XPS spectra were recorded between 0 and 1400 eV in electron binding energy (EBE) scale. A superconducting quantum interference device (SQUID) magnetometer (Quantum Design, MPMS-7) was used to study magnetic properties of ultra small Gd(III) doped iron oxide nanoparticles. Both magnetization (M) versus applied field (H) (i.e., M - H) curves ($-5 \leq H \leq 5$ tesla) at temperature (T) of 300 K and zero-field-cooled (ZFC) M - T curves ($5 \leq T \leq 330$ K) at $H = 100$ oersted (Oe) were recorded. Exactly weighed powder samples were loaded inside nonmagnetic gelatin capsules. Net magnetizations of ultra small Gd(III) doped iron oxide nanoparticles were obtained by using their net masses estimated from TGA curves. Both R_1

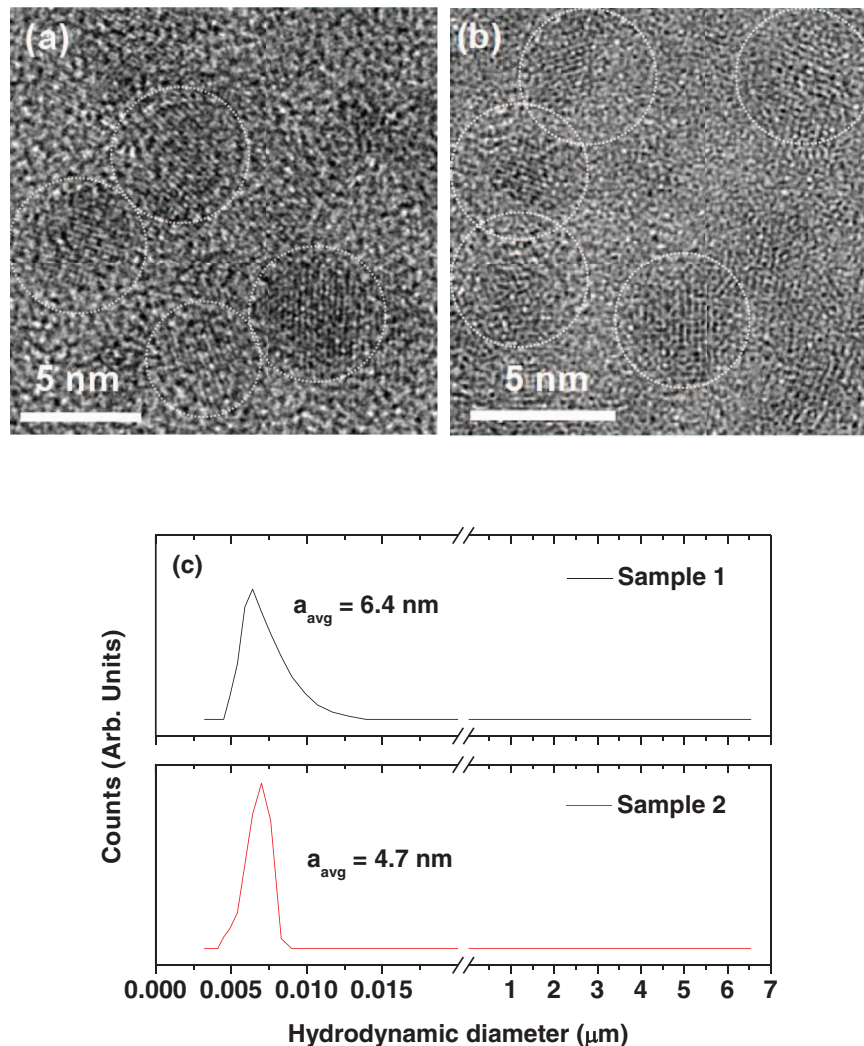


FIG. 1. HRTEM images ((a) sample 1 and (b) sample 2) and (c) DLS patterns of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles. Dotted circles in HRTEM images indicate selected ultra small Gd(III) doped iron oxide nanoparticles.

and R_2 map images as well as both T_1 and T_2 relaxation times were measured by using a 1.5 tesla MRI instrument (GE medical system, GE 1.5 tesla Signa Advantage) equipped with the knee coil (EXTREM). A series of five aqueous sample solutions were prepared at different concentrations (1, 0.5, 0.25, 0.125 and 0.0625 mM Fe). Then, both map images and relaxation times were measured by using these solutions. r_1 and r_2 values were then estimated from the slopes in the plots of $1/T_1$ ($=R_1$) and $1/T_2$ ($=R_2$) as a function of Fe concentration, respectively. The measurement parameters used were as follows: the external MR field (H) = 1.5 tesla, the temperature = 22 °C, the number of acquisition (NEX) = 1, the field of view (FOV) = 16 cm, the phase FOV = 1, the matrix size = 512×512 , the slice thickness = 5 mm, the spacing gap = 0, and the pixel bandwidth = 61.0547, the repetition time (TR) = 2009 ms, and the time to echo (TE) = 9 ms.

III. RESULTS AND DISCUSSION

A. Particle diameter, hydrodynamic diameter, and crystal structure

Figure 1 shows HRTEM images and DLS patterns of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles. Average particle diameters (d_{avg}) of ultra small Gd(III) doped iron

TABLE I. Average particle diameter (d_{avg}), magnetization (M) at 5 tesla and 300 K, and water proton relaxivities (r_1 and r_2) at 1.5 tesla and 22 °C of various nanoparticles.

Sample	Ligand	d_{avg} (nm)	M (emu/g)	r_1 (s ⁻¹ mM ⁻¹)	r_2 (s ⁻¹ mM ⁻¹)	Ref.
Fe ₃ O ₄ /γ-Fe ₂ O ₃	PEG diacid	1.7	6.5	4.46	15.01	17
Sample 1	PEG diacid	2.5	2.3	0.57	2.31	This work
Sample 2	PEG diacid	2.1	2.6	0.90	3.20	This work
Gd ₂ O ₃	D-glucuronic acid	1.0	8.4	9.9	10.5	18

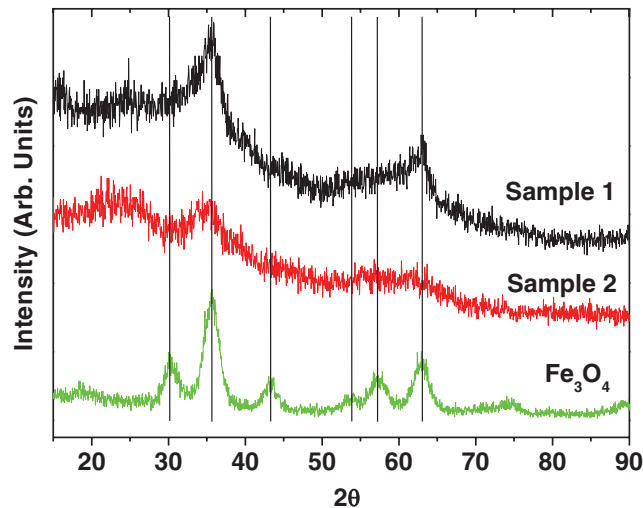


FIG. 2. XRD patterns of two powder samples of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles. An XRD pattern of Fe₃O₄ nanoparticles was given as a reference.

oxide nanoparticles were estimated to be 2.5 and 2.1 nm for samples 1 and 2, respectively, and also provided in Table I. Average hydrodynamic diameters (a_{avg}) were estimated to be 6.4 and 4.7 nm for samples 1 and 2, respectively. XRD patterns indicated that the crystal structure of ultra small Gd(III) ion doped iron oxide nanoparticles closely corresponded to a face-centered cubic (fcc)¹⁹ of Fe₃O₄ as shown in Figure 2. The broad peaks were likely due to ultra small particle diameters.²⁰

B. Surface coating

Surface coating of ultra small Gd(III) doped iron oxide nanoparticles with PEG diacid was investigated by taking FT-IR absorption spectra as shown in Figure 3. Functional groups of PEG diacid such as the C=O stretch at 1590 and 1720 cm⁻¹ and the C-H stretch at 2890 cm⁻¹ confirmed the surface coating of ultra small Gd(III) doped iron oxide nanoparticles with PEG diacid. The PEG diacid has two -COOH groups per molecule. The stretch at 1590 cm⁻¹ is due to -COOH conjugated to a nanoparticle whereas that at 1720 cm⁻¹ is due to free, i.e., unconjugated -COOH. Red shift by ~130 cm⁻¹ in conjugated -COOH had been also observed in various metal oxide nanoparticles,²¹⁻²⁴ supporting our result. The weight percents of surface coated PEG diacid were estimated to be 53 and 66% for samples 1 and 2, respectively, from TGA curves as shown in Figure 4. These indicate that ultra small Gd(III) doped iron oxide nanoparticles were sufficiently coated with PEG diacid, which is necessary for water solubility. The remaining weight percents, i.e., 47 and 34% corresponded to net masses of ultra small Gd(III) doped iron oxide nanoparticles in samples 1 and 2, respectively.

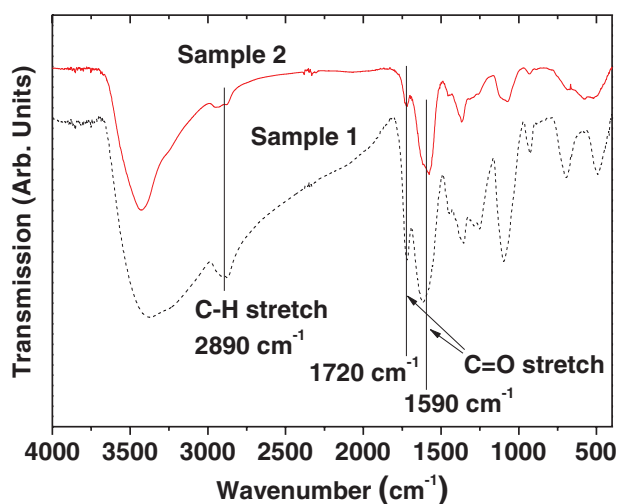


FIG. 3. FT-IR absorption spectra of two powder samples of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles.

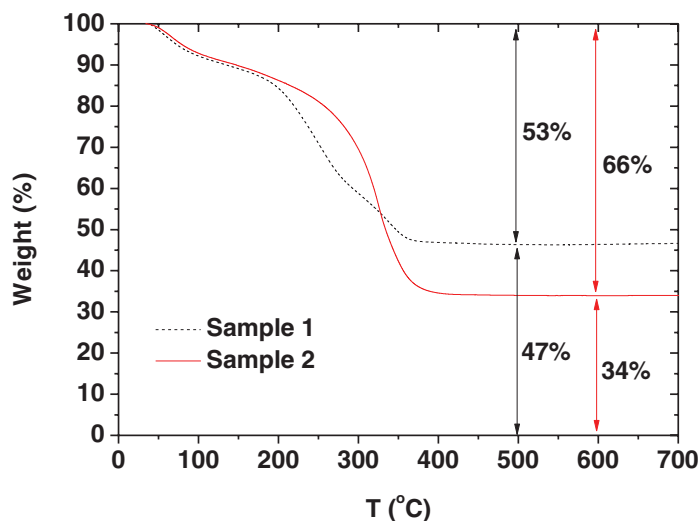


FIG. 4. TGA curves of two powder samples of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles. The top and bottom values indicate weight percents of surface coated PEG diacid and ultra small Gd(III) doped iron oxide nanoparticles in powder samples, respectively.

C. Doping mole percent and surface composition in nanoparticles

Note that Gd(III) precursor mole percents used with respect to those of Fe(III) precursor in reaction were 1 and 10% for samples 1 and 2, respectively, as described in experimental section. However, Gd(III) doping mole percents in nanoparticles estimated from ICPAES analysis were 0.2 and 0.4% for samples 1 and 2, respectively. We attributed this large difference between precursor and doped Gd(III) mole percents as due to mismatch in ionic radius (i.e., $a(\text{Fe(III)})/a(\text{Gd(III)}) = 0.58\text{--}0.74$).²⁵ However, stoichiometric $\text{Gd}_x\text{Fe}_y\text{O}_z$ oxides such as GdFeO_3 , GdFe_2O_4 , Gd_3FeO_6 , and $\text{Gd}_3\text{Fe}_5\text{O}_{12}$ had been prepared at high temperature syntheses through annealing ($>500^\circ\text{C}$).^{10,11,26–38} Therefore, high temperature synthesis should be used to increase Gd(III) doping mole percents.

XPS spectra were taken to investigate surface composition in nanoparticles as shown in Figure 5. As expected from ICPAES analysis, no or negligible signals from Gd (4d) were observed at 143.8 eV ³⁹ within an experimental error limit whereas prominent signals from Fe ($2p_{3/2}$ and $2p_{1/2}$) were observed at 710.9 and 724.5 eV ,³⁹ respectively, in both samples.

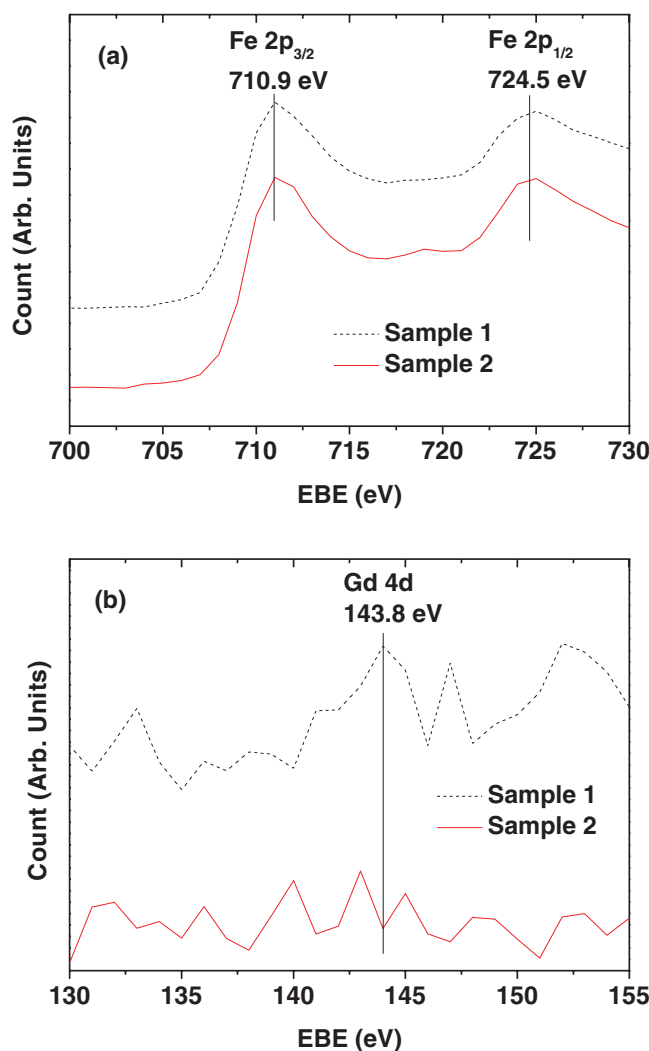


FIG. 5. XPS spectra of two powder samples of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles: (a) Fe (2p_{3/2} and 2p_{1/2}) and (b) Gd (4d).

D. Magnetic properties

Magnetic properties were investigated by recording both M–H curves at $T = 300$ K and ZFC M–T curves at $H = 100$ Oe. Net magnetizations of ultra small Gd(III) doped iron oxide nanoparticles were obtained by using net masses of ultra small Gd(III) doped iron oxide nanoparticles in samples estimated from TGA curves. M–H and ZFC M–T curves are shown in Figures 6(a) and 6(b), respectively. Net magnetizations estimated at $H = 5$ tesla are provided in Table I. These magnetizations are smaller than that ($=84$ emu/g)⁴⁰ of bulk Fe₃O₄ because M–H curves were not fully saturated. An unsaturated M–H curve similar to ours was also observed for the spinel iron oxide nanoparticles with a mean particle diameter of 2.06 nm at room temperature by Yaacob *et al.*⁴¹ They attributed this to a magnetically anomalous surface layer.

Carpenter suggested that the slope at the near zero field in a M–H curve for superparamagnetic nanoparticles with a size distribution should come from the largest nanoparticle.⁴² He suggested that $d_{\text{mag}} = (18kT(dM/dH)_{H=0} / \pi\rho M_s)^{1/3}$ in which the d_{mag} is the maximum magnetic diameter, k the Boltzmann constant ($=1.381 \times 10^{-16}$ erg/K), T the temperature ($=300$ K), ρ the density of nanoparticles (g/cm³), and $(dM/dH)_{H=0}$ the slope near the zero field. By using the estimated $(dM/dH)_{H=0}$ of 9.13617×10^{-5} and 5.6106×10^{-5} emu/gOe for samples 1 and 2, respectively, the saturation magnetization of 84 emu/g for bulk Fe₃O₄,⁴⁰ and the ρ ($=5.1$ g/cm³) of bulk material

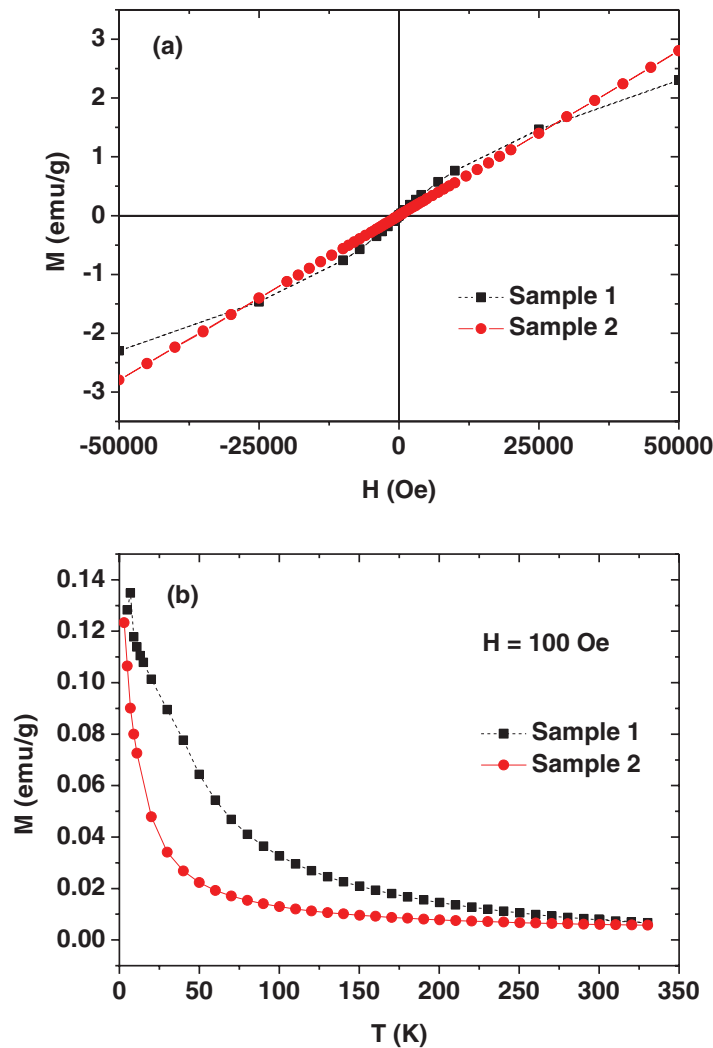


FIG. 6. (a) M-H curves at $T = 300$ K and (b) ZFC M-T curves at $H = 100$ Oe of two powder samples of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles. M corresponds to net M of ultra small Gd(III) doped iron oxide nanoparticles in powder samples.

of Fe_3O_4 ,⁴³ the d_{mag} values were estimated to be 3.6 and 3.1 nm for samples 1 and 2, respectively. These values are slightly larger than average particle diameters of 2.5 and 2.1 nm estimated from HRTEM images, respectively. These are consistent with expectations because the d_{mag} corresponds to the largest nanoparticle in a size distribution.

As provided in Table I, net magnetizations in both samples were smaller than 6.5 and 8.4 emu/g of ultra small iron oxide nanoparticles¹⁷ and ultra small gadolinium oxide nanoparticles,¹⁸ respectively. Therefore, magnetization was reduced by doping Gd(III) into ultra small iron oxide nanoparticles. Reduced magnetizations in various Gd(III) substituted iron oxides had been also observed,^{9–13} supporting our result. It was suggested in those works that Gd(III) opposed the net magnetic moment of Fe(III)/Fe(II) in oxides. An antiferromagnetic coupling between Gd and Fe in metallic thin films had been also reported,^{14–16} supporting our result.

E. Water proton relaxivity and map image

Figure 7 shows plots of $1/T_1$ ($=R_1$) and $1/T_2$ ($=R_2$) inverse relaxation times (=relaxation rates) as a function of Fe concentration. The slopes correspond to r_1 and r_2 relaxivities, respectively.

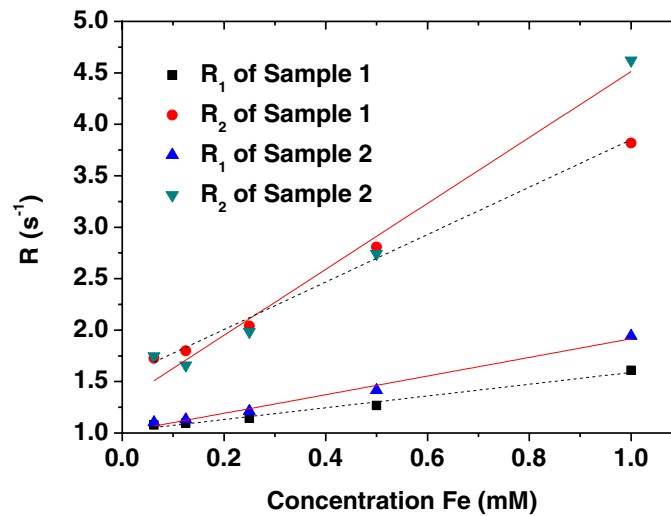


FIG. 7. Plots of R_1 ($=1/T_1$) and R_2 ($=1/T_2$) relaxation rates as a function of Fe concentration of two aqueous solution samples of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles. The slopes correspond to r_1 and r_2 relaxivities, respectively.

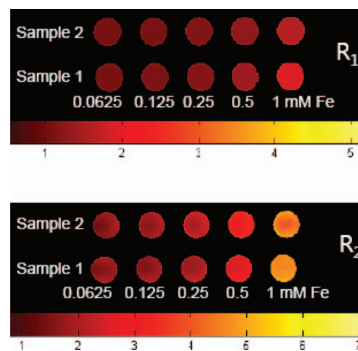


FIG. 8. R_1 and R_2 map images of aqueous sample solutions of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles as a function of Fe concentration.

Table I lists relaxivities of various nanoparticles including PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles. R_1 and R_2 map images are presented in Figure 8. As expected from r_1 and r_2 values, dose-dependent contrast enhancements in R_1 map images were negligible and those in R_2 map images were only decent, implying that PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles are not suitable for both T_1 and T_2 MRI contrast agents at clinical MR fields.

Figure 9 clearly shows Gd(III) doping effect on magnetization and r_1 and r_2 values. That is, magnetizations and r_1 and r_2 values of two samples are smaller than those of ultra small iron oxide nanoparticles¹⁷ and ultra small gadolinium oxide nanoparticles.¹⁸ As investigated by others, magnetic moment of Gd(III) is anti-parallel to net magnetic moment of Fe(III)/Fe(II) in oxides.^{9–13} Therefore, reduced magnetizations in both samples are due to this. Note that r_1 and r_2 values are proportional to square of magnetization.⁴⁴ Therefore, reduced r_1 and r_2 values are due to reduced magnetizations in both samples.

IV. CONCLUSIONS

In summary we for the first time investigated Gd(III) doping effect on magnetization and water proton relaxivities by synthesizing two samples of ultra small Gd(III) doped iron oxide nanoparticles. Gd(III) doping mole percents were 0.2 and 0.4% for samples 1 and 2, respectively. Average particle

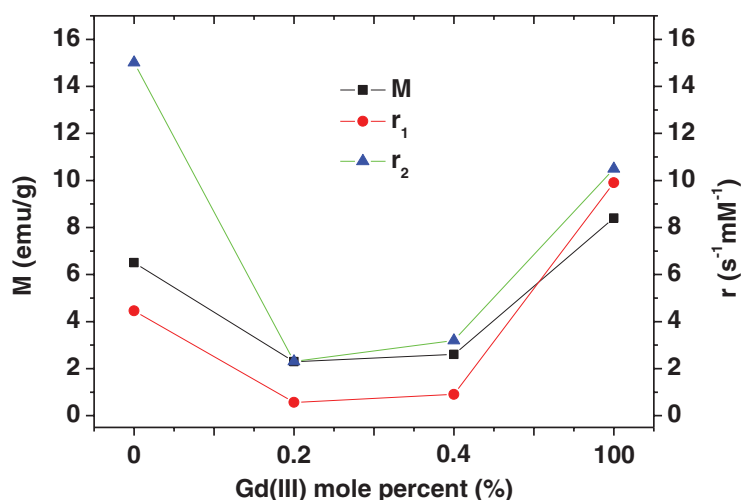


FIG. 9. Plots of magnetization (M) and water proton relaxivities (r_1 and r_2) of two samples of PEG diacid coated ultra small Gd(III) doped iron oxide nanoparticles as a function of Gd(III) mole percent.

diameters were 2.5 and 2.1 nm for samples 1 and 2, respectively. Although it was expected that doping of high spin Gd(III) ($S = 7/2$) into ultra small iron oxide nanoparticles would increase water proton relaxivities (especially, r_1 values), reduced r_1 and r_2 values were observed in both samples. This result was ascribed to opposing effect of Gd(III) on net magnetic moment of Fe(III)/Fe(II) in oxides, giving reduced magnetizations and as a result reduced r_1 and r_2 values.

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