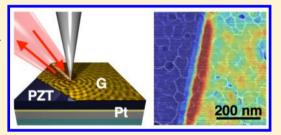


Tuning and Persistent Switching of Graphene Plasmons on a Ferroelectric Substrate

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Supporting Information

ABSTRACT: We characterized plasmon propagation in graphene on thin films of the high-κ dielectric PbZr_{0.3}Ti_{0.7}O₃ (PZT). Significant modulation (up to $\pm 75\%$) of the plasmon wavelength was achieved with application of ultrasmall voltages ($< \pm 1$ V) across PZT. Analysis of the observed plasmonic fringes at the graphene edge indicates that carriers in graphene on PZT behave as noninteracting Dirac Fermions approximated by a semiclassical Drude response, which may be attributed to strong dielectric screening at the graphene/PZT interface. Additionally, significant plasmon scattering occurs at the grain boundaries of PZT from topographic and/or



polarization induced graphene conductivity variation in the interior of graphene, reducing the overall plasmon propagation length. Lastly, through application of 2 V across PZT, we demonstrate the capability to persistently modify the plasmonic response of graphene through transient voltage application.

KEYWORDS: graphene, ferroelectric, PZT, near-field microscopy, plasmonics, s-SNOM

raphene plasmonics is a burgeoning field of study that has I attracted significant interest due to graphene's unique electronic and optical properties, as well as the ability to conveniently modify these properties through back-gating or optical pumping. 1-5 Extensive work has been performed to characterize the capabilities of different gate dielectrics for tuning of the electronic properties of graphene. Among the most studied gate dielectrics are SiO2, for its robustness with voltage application, and hexagonal boron nitride (hBN), for the high mobility it lends to graphene microcrystals or films.⁶⁻⁸ However, the realization of practical plasmonic devices requires implementation of alternative dielectrics to achieve larger variance in graphene properties and to potentially attain additional control mechanisms (e.g., persistent effects). Ferroelectrics are one intriguing option. In general, such dielectrics have a large DC permittivity (κ or $\varepsilon_{\rm DC}$) in the linear, low-field regime and also may enable persistent tuning in the high-field nonlinear regime. 9-14 As a result, ferroelectrics may find applications in graphene-based electronics and plasmonics. Although previous experimental works on ferroelectric modification of graphene primarily investigated control of electronic properties, 9-19 we focus here on the plasmonic and photonic response. We examined the potential for tuning of the plasmonic response of graphene through the use of one such dielectric, PbZr_{0.3}Ti_{0.7}O₃ (PZT). The dielectric constant of PZT can reach values of several hundred or thousand, many times that of SiO₂, making it an extremely effective insulator for

back-gated structures. 10 Therefore, PZT's potency for use in future plasmonic devices warrants investigation.

To characterize graphene plasmon propagation on PZT substrates, we employed scattering-type scanning near field optical microscopy (s-SNOM).^{20–22} In these measurements, infrared light is focused at the sharp apex of a metallic atomic force microscope (AFM) tip as seen in Figure 1a (Neaspec).^{23–25} A fraction of the incident radiation polarizes the tip and is confined to length scales on the order of the tip diameter (~20 nm) where the tip interacts with the sample. As a result, the light scattered from the tip-sample system contains information on the sample properties at deeply subdiffraction limited length scales.²⁴ Additionally, through the enhanced range of momenta excited by the tip, 5,26,27 it is possible to overcome the momentum mismatch between plasmonic modes and the incident far field light allowing the tip to launch radially propagating plasmons, which are reflected from topographic and electronic features in the graphene (e.g., edges, wrinkles, folds, grain boundaries) as shown schematically in Figure 1a.²³ These reflected waves create oscillations in the local electric field directly beneath the tip corresponding to a detected variation in near-field signal. Analysis of the oscillating near field signal yields information including the plasmon wave-

Received: January 11, 2015 Revised: July 9, 2015 Published: July 16, 2015

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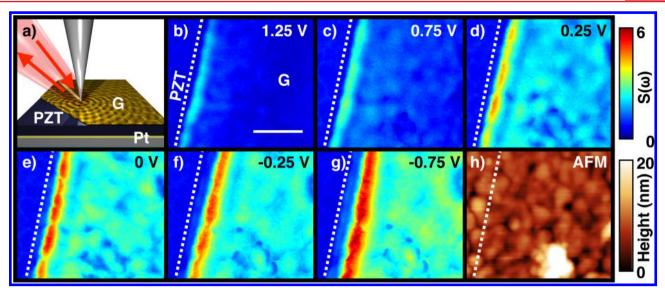


Figure 1. (a) Schematic of our nanoimaging setup. The concentric yellow circles represent propagating plasmonic waves launched by the tip and reflected by the graphene edge. (b–g) Normalized voltage dependent near-field amplitude $(S(\omega))$ of graphene (right) on PZT (left) taken at $\omega = 890 \text{ cm}^{-1}$. The dotted white line indicates the position of the graphene (denoted as G) edge determined from AFM. A strong plasmonic fringe is visible near this edge. Fringe brightness and width increase with decreasing voltage. (h) AFM image, taken simultaneously with the 0 V data, showing device roughness in the same region as the SNOM images shown in (b-g).

length (λ_p) given by twice its periodicity as seen in Figure 2, and corresponding damping coefficient, γ_p . Both these parameters are directly related to the properties of graphene, including the scattering rate and carrier density.^{5,27}

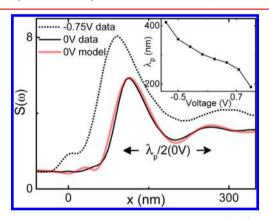


Figure 2. Comparison of simulated plasmonic profile (red) and data (solid black) taken from Figure 1d with 0 V applied to the back gate. There is good agreement between the two profiles for $\gamma_p=0.13$. The profile for the longest wavelength data (dotted black) is also shown. (Inset) Gate-dependent plasmonic wavelength obtained from the images in Figure 1.

Our device consists of a graphene microcrystals exfoliated on a 360 nm-thick polycrystalline PZT film grown via sol gel on a 50 nm platinum layer that functions as our back-gate, as shown in the schematic in Figure 1a. This back-gated structure allows for straightforward carrier doping of graphene in the linear dielectric regime of PZT ($-1~{\rm V} < V_{\rm G} < +1~{\rm V}$) during near-field measurements. To accurately probe graphene plasmons without the influence of plasmon–phonon coupling, which modifies the frequency–momentum dispersion of the graphene plasmon from the simple square-root dependence, 26 the sample was probed using a coherent infrared beam generated by a ${\rm CO}_2$ laser at $\omega=890~{\rm cm}^{-1}$, where no PZT phonons are present. We first characterized our device at back-gate voltages ranging from

 $V_{\rm G}=-0.75~{
m V}-1.25~{
m V}$, where negative and positive voltages indicate an increase or decrease in the hole concentration in the graphene. The resulting normalized near field amplitudes are shown in Figure 1b–g. In these images, the near field signal is related to the local z component of the electric field (E_z) beneath the tip, with higher signal generally relating to a higher E_z in graphene. An AFM image of the same region is shown in Figure 1h.

We first survey the gross features present in the near field images. The most prominent aspect of these images is the bright fringe seen at the edge (white dotted line across each image) of the graphene flake. According to previous studies, 5,27 they are formed due to interference between tip-launched and edge reflected plasmons. As the hole concentration (n), and therefore the graphene conductivity (σ) , increases at more negative V_{G} , the width and intensity of these fringes, along with the overall graphene signal, monotonically grow in congruence with previous measurements of structures utilizing SiO2 gate insulator. This monotonic change in n with $V_{\rm G}$ reveals that the Fermi energy $(E_{\rm F})$ never crosses the charge neutrality point (CNP), which would correspond to a reduction and subsequent enhancement of n. Rather, this result suggests a positive carrier sign at all gate voltages in this range and indicates unintentional hole doping of the graphene at 0 V. From near-field measurements over a broader voltage range, we expect that the voltage at the CNP is between 1.0 and 1.5 V (see Supporting Information). This is consistent with a minimal carrier concentration at $V_{\rm G}$ = 1.25 V, resulting in the near invisibility of graphene in Figure 1b where only a weak fringe is visible, as charge neutral graphene is nearly infrared transparent, transmitting approximately 98% of incident light. The existence of a weak fringe in spite of low carrier concentration is likely due to edge doping, as was previously seen in graphene on $\mathrm{SiO_{2}}^{28,29}$ and is also reflected in the decreasing overall signal in regions of graphene away from the edge.

To directly characterize plasmon behavior, we examined the near field signal close to the graphene edge where interference fringes are most prominent. The plasmon wavelength, λ_p , can

be directly extracted from the images in Figure 1b-g by obtaining a line profile of our SNOM signal perpendicular to the edge. Many such profiles are obtained and averaged to eliminate artifacts from variation in substrate topography and substrate optical properties. The resulting profile obtained from our 0 V data is plotted as the solid black curve in Figure 2. This profile shows characteristic fringes, which decrease in amplitude with increasing distance from the graphene edge. Modification of the applied voltage results in variation of the amplitude and wavelength of the interference fringes. The values of $\lambda_{\rm p}$ extracted from the averaged line profiles extracted at each applied voltage are shown in the inset of Figure 2. The measured wavelength varies by as much as $\pm 75\%$ of its value at 0 V with a voltage range of only ± 1 V. This is more than 2 orders of magnitude greater than what is seen in SiO₂supported graphene structures for the same range of applied electric fields.

The measurements reported in Figures 1 and 2 can serve as an alternative method for directly determining the DC dielectric constant of PZT. The theoretical plasmon dispersion can be calculated from the maximum of the reflection coefficient, $R_{\rm p}(\omega=890~{\rm cm}^{-1})$, which has a well-defined dependence on the graphene carrier concentration. A comparison of this theoretical dispersion and the measured plasmon wavelength allows for extraction of the graphene carrier concentration, n, at each applied voltage, $V_{\rm G}$. This carrier density then yields the dielectric constant of PZT: $\varepsilon_{\rm DC}=(ned/l\ V_{\rm G}-V_{\rm CNP}|)$ where e is the electron charge, d is the PZT thickness, and $V_{\rm CNP}=1.3~{\rm V}$ consistent with our near field measurements. This analysis yields a range of values $\varepsilon_{\rm DC}=450-550$, well within the expected range for PZT and in agreement with direct measurements on similar films.

In spite of the large wavelengths we observe in Figures 1 and 2, the plasmon propagation length appears limited to ~500 nm even for the longest $\lambda_p = 420$ nm (black dotted line in Figure 2), indicating the presence of plasmonic scattering in these devices. Information on losses and scattering can be obtained by examining the decay in the interference fringe strength with increasing tip-to-edge distance. Although averaging many curves with slight variation in phase has the potential to mask intrinsic damping, a comparison of damping extracted from averaged and nonaveraged data shows little variation in the plasmonic damping. To obtain the plasmon damping coefficient, γ_p , we modeled our plasmonic profile obtained at $V_{\rm G} = 0$ V using the procedure outlined in Fei et al., approximating the tip as an elongated ellipsoid.³² The value of $\gamma_{\rm p}$ is varied until the theoretical fringe profile matches experimental data. The resulting simulated profile is shown by the red curve in Figure 2 and fits our experimental data (solid black) most closely for $\gamma_p = 0.135$. The measured damping and wavelength are directly related to the plasmon dispersion via $\lambda_{\rm p}=2\pi/q_1$ and $\gamma_{\rm p}=q_2/q_1$ and complex graphene conductivity, $\sigma = \sigma_1 + i\sigma_2$, with the theoretical dispersion given

$$q_{\rm p} = q_1 + iq_2 = \frac{i\omega[1 + \epsilon(\omega)]}{4\pi\sigma(\omega)} \tag{1}$$

where $\omega=890~{\rm cm}^{-1}$ is the excitation frequency and $\epsilon(890~{\rm cm}^{-1})=2.58+0.12i$ is the ellipsometrically measured permittivity of the PZT at our probing frequency. At $V_{\rm G}=0$ V, the measured plasmon wavelength ($\lambda_{\rm p}=270~{\rm nm}$) and damping ($\gamma_{\rm p}=0.135$) yield a graphene conductivity of $\sigma(890$

cm⁻¹) = $(0.56 + 5.8i) \times e^2/h$. The uncertainty in the extracted plasmon wavelength is $\sim \pm 30$ nm and is a result of several factors including the tip radius (20 nm), which determines the resolution of our measurements. The conductivity, determined from near field data, can be compared to a semiclassical Drude model for graphene using a carrier relaxation rate, $\tau^{-1} = 70$ cm⁻¹, determined independently from the DC mobility of $\mu = 2000$ cm²/(V·s) measured using transport in single grains of chemical vapor deposition grown graphene. The conductivity from this model is given by³³

$$\sigma(\omega) = \frac{e^2 E_F}{\pi \hbar} \frac{i}{\omega + i\tau^{-1}} = (0.5 + 6.7i) \times e^2/h$$
 (2)

where $E_{\rm F}(0~{\rm V})=3000~{\rm cm}^{-1}$ is the graphene Fermi energy. Plasmonic damping originates from a combination of losses within graphene itself and those resulting from the dielectric environment around graphene as given by $\gamma_{\rm p} \approx \sigma_{\rm l}/\sigma_{\rm 2}$ + ($\varepsilon_{\rm 2}/(1$ + ε_1) with $\sigma_1/\sigma_2 = 0.098$ and $(\varepsilon_2/(1 + \varepsilon_1)) = 0.037$ giving the relative contributions from each channel. Additionally, the agreement between the noninteracting Drude model and our measured conductivity indicates that damping in the graphene can be ascribed to the high scattering rate of the electron gas in graphene on PZT. In this regime of enhanced scattering, the many body interactions that were previously reported for graphene on ${\rm SiO_2}^5$ and ${\rm BN}^{34}$ do not play a significant role. Additionally, the absence of many body effects in our device can be understood as a direct result of the high dielectric constant of PZT, which serves to screen electron-electron interaction reducing the overall strength of the Coulomb force by several orders of magnitude as compared to structures using SiO₂ as the gate insulator.³⁵

Surprisingly, the value of the loss factor γ_n observed in our device is comparable to that measured by Fei et al.⁵ However, in contrast to their results, it is difficult to distinguish fringes beyond the first directly in our imaging data. Examination of the internal regions of graphene shows the existence of bright fringes that are reminiscent of the plasmonic interference patterns observed at the edges of graphene flakes. These findings suggest the existence of plasmonic reflectors and scatterers within the interior of graphene, as these are a prerequisite for the formation of these additional interference patterns.³⁶ Initial examination of the data suggests a correlation between the underlying PZT grain topography and reflected plasmons. The AFM image in Figure 1h shows a similar topographic landscape on bare PZT and graphene covered PZT, indicating that graphene sits closely to the surface of PZT, directly following the grain structure below it. The surface roughness of PZT is \sim 4 nm across the 1 \times 1 μ m region shown in Figure 1h with similar roughness reflected in the graphene. Individual PZT grains are visible in topography within both regions, with dimensions on the order of tens of nanometers.

To verify PZT grain boundaries as the source of plasmonic reflection within the interior of graphene, conventional Laplacian edge detection techniques were applied to the AFM data in Figure 1h resulting in the processed image in Figure 3a. The white traces running throughout this panel indicate the edges of PZT grains. In order to determine the relationship between the bright fringes in the graphene interior and the PZT grains, we overlaid the near field image from Figure 1g on top of the processed AFM image to create Figure 3b. In this image, the PZT grain edges are simultaneously visible with the near field signal. The bright features in the

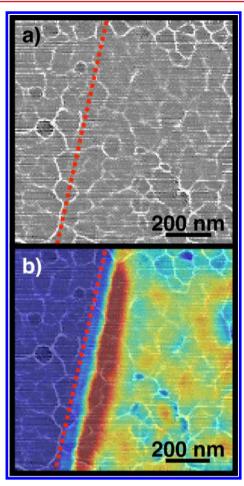


Figure 3. (a) Image obtained from conventional Laplacian edge detection techniques performed on topography from Figure 1h. Grain borders shown by white lines throughout image. (b) Near field signal from Figure 1g overlaid on the processed topography shown in (a).

interior of graphene appear to be enclosed within PZT grains suggesting that plasmons are reflected at the boundaries of these grains. This points to two possible sources of plasmonic reflection within the interior of graphene, both of which are governed by the existence of PZT grains: (1) topographic variation created in graphene by PZT creates strong enough variation in local graphene conductivity (from strain or other effects) to reflect plasmons and (2) different grains in PZT are polarized differently causing local conductivity changes which reflect plasmons. These two sources are difficult to distinguish with our current measurements, requiring alternative techniques, but do not modify the analysis described here.

For such analysis to hold, trivial topographical influence on near field signal³⁹ must be accounted for as topographic variation can modify the collected near field signal. In general, concave surface features create higher near field signal resulting from field enhancement, whereas convex surfaces result in lower signal. This is in contrast to what we observe with higher signal concentrated in the central convex regions of the PZT grains. Our analysis, therefore, suggests that PZT created graphene conductivity variation is a significant source of plasmonic reflection within the interior of graphene. Scattering from these local conductivity changes exacerbates the task of distinguishing multiple fringes close to the edge in the near field images without averaged profiles as the additional reflectors can deflect propagating plasmons before they are able to make the

round trip from the tip to the edge and back. The observation of rapid plasmon decoherence in our near field images, compared with that in SiO_2 and resulting from scattering from PZT grains, suggests that significant improvement in the observed propagation length may be achieved by using graphene on atomically flat epitaxial PZT giving plasmons a smooth surface across which they can propagate. ¹²

In spite of the significant scattering resulting from topographic features, we were able to employ ferroelectric switching of PZT for persistent tuning of graphene plasmons. As a ferroelectric material, PZT can be persistently polarized with application of transient electric fields greater than $\sim\!30~\rm kV/cm$ (1 V across our device). It was previously demonstrated that this field application modifies the voltage of the graphene CNP, producing a corresponding change in the graphene carrier concentration and plasmonic characteristics, even after removal of the voltage. 9,10,13,40

To test the possibility of plasmonic memory gating, we first obtained a near field image of an unpolarized region of the device at 0 V. After this measurement, a gate voltage of 2 V was applied for several minutes to polarize the PZT. Although the fairly low field from application of 2 V to our PZT film is not sufficient for complete polarization, it is larger than the coercive field of PZT and should therefore be enough to partially and persistently align the dipoles within the film (see Supporting Information). Following voltage removal, a second image was obtained in the same region of the device. Because the application of positive voltages reduces the hole concentration, any film polarization remaining after voltage application should result in a similar decrease in near field signal as that seen with increasingly positive $V_{\rm G}$ in our continuous bias application measurements in Figure 1b-g. The two near field images taken before and after memory gating, and their corresponding profiles, are shown in Figure 4a-d. Prior to voltage application,

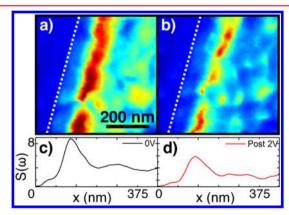


Figure 4. (a) Normalized near field amplitude of graphene on an unpolarized region of PZT. (b) Second image taken in the same region as (a) after application and removal of 2 V across PZT. (c) and (d) Line profiles corresponding to the plasmonic fringe seen in (a) and (b), respectively.

a bright fringe is visible in Figure 4a with multiple maxima seen in the corresponding line profile shown in Figure 4c. The equivalent data taken after voltage removal, Figure 4b and d, respectively, demonstrate significant reduction in fringe strength and overall graphene brightness, as well as the disappearance of the second fringe in the line profile. These changes are consistent with a reduction in the magnitude of graphene doping, a quantity to which our near-field data is

extremely sensitive. Thus, these data demonstrate the ability to persistently turn off the plasmonic response with transient voltage application. These changes to graphene carrier concentration were reversible with negative bias application and faded with extended 0 V bias application as seen in other ferroelectric based devices. 41 This time dependent relaxation is consistent with the following picture of partial PZT polarization. Initial application of 2 V begins to reorient the polarization of each PZT grain into the out of plane direction. With removal of the driving voltage, the polarization directions of the grains relax back to their original disordered lower energy state. This mechanism can be further verified through the implementation of simultaneous piezoresponse force microscopy and plasmonic near-field studies. With position dependent control of PZT polarization, as has been demonstrated using AFM, 42,43 such plasmonic memory gating could be employed for the creation of reconfigurable plasmonic nanocircuits without the need for continuous stimulus.

Our results demonstrate the effectiveness of PZT as a gate dielectric for control of graphene plasmons. The dramatic increase in tunability, arising from the large dielectric constant of PZT, minimizes the voltages necessary for significant modification of plasmonic response. Additionally, the potential for persistent modification of plasmonic properties with polarization of ferroelectric films presents the opportunity for creation of a variety of graphene based nanoplasmonic elements (circuits, reflectors, and lenses). Although losses are present in our device, exfoliation of graphene on atomically flat epitaxial PZT^{44–46} could dramatically reduce unwanted plasmonic scattering, making PZT an ideal substrate for low voltage graphene based electronic applications.

ASSOCIATED CONTENT

Supporting Information

Electrical transport measurements on similar devices. Probe of hysteresis in graphene near-field signal. The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.nanolett.5b00125.

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Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

This work is supported by ONR (N00014-13-1-0464). Instrument development at UCSD is supported by DOE-BES. B.Ö. acknowledges support by the NUS Young Investigator Award and the SMF-NUS Research Horizons Award 2009-Phase II. A.H.C.N. acknowledges the National Research Foundation, Prime Minister's Office, Singapore, under its Medium Sized Centre Programme and CRP award "Novel 2D materials with tailored properties: beyond graphene" (R-144-000-295-281).

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