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Fluorination, and Tunneling across Molecular Junctions

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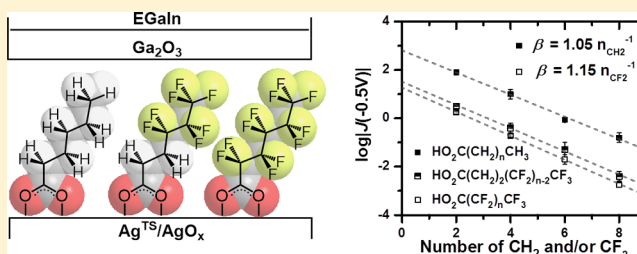
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Supporting Information

ABSTRACT: This paper describes the influence of the substitution of fluorine for hydrogen on the rate of charge transport by hole tunneling through junctions of the form $\text{Ag}^{\text{TS}}\text{O}_2\text{C}(\text{CH}_2)_n(\text{CF}_2)_m\text{T}/\text{Ga}_2\text{O}_3/\text{EGaIn}$, where T is methyl (CH_3) or trifluoromethyl (CF_3). Alkanoate-based self-assembled monolayers (SAMs) having perfluorinated groups (R^{F}) show current densities that are lower (by factors of 20–30) than those of the homologous hydrocarbons (R^{H}), while the attenuation factors of the simplified Simmons equation for methylene ($\beta = (1.05 \pm 0.02)n_{\text{CH}_2}^{-1}$) and difluoromethylene ($\beta = (1.15 \pm 0.02)n_{\text{CF}_2}^{-1}$) are similar (although the value for $(\text{CF}_2)_n$ is statistically significantly larger). A comparative study focusing on the terminal fluorine substituents in SAMs of ω -tolyl- and -phenyl-alkanoates suggests that the C–F// Ga_2O_3 interface is responsible for the lower tunneling currents for CF_3 . The decrease in the rate of charge transport in SAMs with R^{F} groups (relative to homologous R^{H} groups) is plausibly due to an increase in the height of the tunneling barrier at the T// Ga_2O_3 interface, and/or to weak van der Waals interactions at that interface.



INTRODUCTION

Studies of charge tunneling through metal–molecule–metal (MMM) junctions have focused predominately on testing hypotheses that correlate the chemical and electronic structure of the molecules with current densities (or in the case of single-molecule studies, with currents). A convenient, semiquantitative theoretical framework around which to organize trends relating measurable parameters (e.g., the length of a $(\text{CH}_2)_n$ group) to experimental data (e.g., current densities at a fixed applied voltage) has been the simplified Simmons equation (eq 1).^{1–11} In this approximation, the tunneling barrier is

$$J(V) = J_0(V) e^{-\beta d} = J_0(V) 10^{-\beta d/2.303} \quad (1)$$

approximated as rectangular, with width d , and a height related to the attenuation factor β .^{12,13} $J(V)$ is current density (A/cm^2) at an applied bias V , and J_0 is loosely interpretable as the injection current for a hypothetical junction with $d = 0$. Changes in the topography of the barrier, the energies of the frontier orbitals, molecular dipoles, and polarizabilities of the insulating molecules in the junctions, are ignored or considered as part of J_0 .^{14–18}

We have studied this type of system using self-assembled monolayer (SAM)-based junctions of the structure Au^{TS} or

$\text{Ag}^{\text{TS}}/\text{A}-\text{R}_1-\text{M}-\text{R}_2-\text{T}/\text{Ga}_2\text{O}_3/\text{EGaIn}$; previous papers describe these studies.^{11,19–25} We have used a variety of polar, aromatic, and aliphatic groups for the “anchoring” (A), “middle” (M), and “terminal” (T) groups. One of the unexpected implications of these studies has been that increasing the strength of the interaction across the T// Ga_2O_3 interface does not seem to increase the tunneling current density of n -alkyl SAMs;²⁰ decreasing this strength does, however, seem to decrease the tunneling current. The topography of these tunneling barriers seems to be dominated by the electronic structure of the insulating alkyl chains. A theoretical study by Nijhuis and Zhang calculated that the T// Ga_2O_3 interface was the highest region in the tunneling barrier.²⁶ One possible, testable hypothesis based on these experimental and theoretical studies might be that decreasing the strength of the T// Ga_2O_3 interface might decrease the tunneling current density by increasing the height of the tunneling barrier at its highest point (i.e., at that interface).

The work we describe here was designed to test this possibility by replacing C–H bonds in the terminal group T with C–F bonds, and comparing tunneling current densities.

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Replacing hydrogen (C–H bonds) with less polarizable fluorine (C–F bonds) often changes the structural, chemical, and electronic properties of hydrocarbons.^{27,28} For example, electronegative fluorine influences the frontier orbital energy of alkanes, lowers their surface energy and polarizability, and disrupts interchain packing and van der Waals interactions between chains in a SAM.^{29,30} We explored the influence of the extent of fluorination of *n*-alkyl SAMs on the rate of charge transport across large-area junctions of the form Ag^{TS}O₂C(CH₂)_{*n*}(CF₂)_{*m*}T//Ga₂O₃/EGaIn (*n*, *m* = 0, 2, 4, 6, 8) at ±0.5 V. We varied the number of methylene (CH₂) and difluoromethylene (CF₂) groups in the backbone of the SAM, and changed the terminal function groups (T) of the SAM from H to F, and from CH₃ to CF₃, in order to test two possibilities: (i) In the backbone of the molecules, CH₂ and CF₂ might contribute differently to the height of the tunneling barrier, since electronegative fluorine substituents could lower the energy of the highest occupied molecular orbital (HOMO) of the *n*-alkyl SAMs, increase β, and decrease the tunneling current density. (ii) In the terminal group T, replacing C–H bonds with C–F bonds might decrease the strength of the van der Waals interaction across the T//Ga₂O₃ interface, raise the tunneling barrier in this region, decrease *J*₀(V), and decrease the tunneling current density.

Replacing C–H bonds with C–F bonds in the terminal group T did, in fact, significantly lower the tunneling current density in a number of these compounds. Our specific focus in this work concerned the mechanism and origin of this effect. Experimentally, we observed that CH₂ and CF₂ contribute similarly (although perhaps distinguishably; see below) to the effective height of the tunneling barrier, and that the C–F//Ga₂O₃ interface is responsible for low tunneling currents of some fluorinated hydrocarbons (e.g., perfluoroalkanes). The values of the attenuation factor, β in eq 1, for CH₂ (β = (1.05 ± 0.02)*n*_{CH₂}^{−1}) and CF₂ (β = (1.15 ± 0.02)*n*_{CF₂}^{−1}) are similar, although distinguishable. The extrapolated current density (at −0.5 V) of the perfluorinated *n*-alkyl SAMs (log|*J*_{CF₃//Ga₂O₃}| = 1.3 ± 0.2 for HO₂C(CF₂)_{*n*}CF₃ when *n* = 0) and the homologous hydrocarbons (log|*J*_{CH₃//Ga₂O₃}| = 2.8 ± 0.2) differ significantly and in a way that suggests that the difference is primarily attributable to the T//Ga₂O₃ interface. We observed similar offsets of *J*₀ for *para*-substituted (T = CH₃ or CF₃) ω-tolyl-alkanoates (HO₂C(CH₂)_{*n*}(C₆H₄)T) and oligophenyl carboxylates (HO₂C(C₆H₄)_{*n*}T). In all of these compounds, the rates of charge transport across the C–F//Ga₂O₃ interface are consistently lower by factors of 25–30 than those across the C–H//Ga₂O₃ interface.

■ BACKGROUND

SAMs with perfluorinated groups (R^F) show structural and electronic properties that are different than those of the homologous hydrocarbons (R^H). The aliphatic C–F bond is ~1.4 Å in length and has a dipole moment of about 1.4 D; the aliphatic C–H bond is ~1.1 Å in length, and has a dipole moment of 0.3 D. The van der Waals radius of fluorine (1.47 Å) is larger than that of hydrogen (1.2 Å).^{31,32} Fluorination of hydrocarbons raises the ionization potential of linear hydrocarbons (fluorocarbons) from 10 to 17 eV,³³ lowers the HOMO energy by 1.0–1.5 eV (and also increases the HOMO–LUMO gap),³⁴ and shifts the UV absorption band toward higher energy (approximately from 122 to 187 nm);³⁵ fluorination also changes the gas solubility, and the boiling and

melting points of hydrocarbons, in ways compatible with weaker intermolecular forces between fluorocarbons than between hydrocarbons.³⁵ Fluorocarbons, which have lower polarizability than hydrocarbons,^{32,34} are functional groups that have been used to manipulate the chemical and electronic properties of metals and metal oxides.^{36–41} SAMs with R^F groups exhibit both hydrophobic and lipophobic properties, and are more thermally stable and chemically inert than SAMs based on analogous hydrocarbons (R^H).^{36,42,43} In contrast to R^H chains (–(CH₂)_{*n*}–), which generally show linear, trans-extended packing in SAMs, the backbone of R^F chains (–(CF₂)_{*n*}–) form a helix-like structure that creates a slightly larger footprint for the individual R^F molecules.^{44–46} Both R^H and R^F molecules form densely packed SAMs.

SAMs have been used to modify the electronic properties of the interface between the electrodes and the organic active layers in organic electronics.⁴¹ For example, Blom et al. demonstrated that SAMs of *n*-alkanethiols can be used to decrease the work functions of gold and silver, whereas SAMs of perfluorinated *n*-alkanethiols can be used to increase these work functions (the R^H and R^F groups introduce opposite dipoles at the surface of the electrodes).⁴⁷ Cho and Tao reported that the work function of silver and aluminum can also be tuned using carboxylate-based SAMs;⁴⁸ depending on the length and the extent of fluorination in the structure of *n*-alkanoates, the work function of SAM-modified silver can be shifted from 4.6 eV (bare Ag) to 5.7 eV (R^F-bound Ag).^{48–50} This work also confirmed that the carboxylate anchoring group binds in a bidentate form to the surface of Ag (where a layer of native silver oxide possibly exists at the interface between the metal and the carboxylate) with an angle of inclination of the R^F chain of ca. 28° to the surface normal.⁴⁸

■ RESULTS AND DISCUSSION

We prepared SAMs on template-stripped silver using commercially available fluorinated *n*-alkanoic acids, following a previously reported procedure (Figure 1a).⁴⁸ SAMs were formed by introducing freshly prepared Ag^{TS} substrates⁵¹ into 1 mM solutions of fluorinated alkanic acids in *n*-hexadecane for 10 min; further immersion (up to 48 h) did not change either the static contact angle of water (~121 ± 5°) on the SAMs, or the yields of working junctions. Following incubation at room temperature (and directly before electrical measurements), we rinsed the SAM-bound Ag substrates three times with anhydrous hexane (1 mL each time) to remove the residual hexadecane from the surface of R^F-bound silver, and dried the substrates under a gentle stream of nitrogen. We measured *J*(V) for junctions of the structure Ag^{TS}O₂C(CH₂)_{*n*}(CF₂)_{*m*}T//Ga₂O₃/EGaIn using what we call “selected” conical EGaIn tips⁴ (with a geometrical contact area of ~1800 μm²) over the range of ±0.5 V as a function of the number of CH₂ and/or CF₂ units (where *n*, *m* = 0, 2, 4, 6, 8 and T = CH₃ or CF₃). We and others have discussed the details of the EGaIn electrode extensively.^{11,52–54} (See the Supporting Information for additional details on the measurement procedure and data analysis.)

Although the structure and surface energy of R^F and R^H SAMs differ significantly, the contributions of CH₂ and CF₂ groups to the height of the tunneling barrier are apparently slightly different in their contributions to the value of β. Figure 1b shows a plot of log|*J*(−0.5 V)| versus the number of CH₂ and/or CF₂ groups for junctions comprising SAMs of 2H,2H,3H,3H-perfluoroalkanoates, perfluoroalkanoates, and

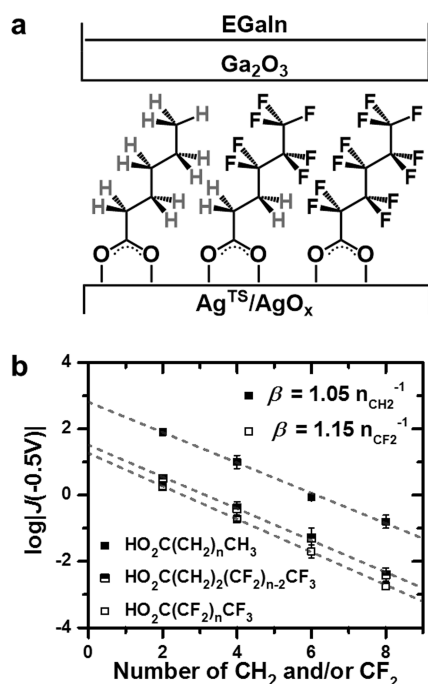


Figure 1. (a) Cartoon representing junctions comprising SAMs of $\text{HO}_2\text{C}(\text{CH}_2)_n\text{CH}_3$, $\text{HO}_2\text{C}(\text{CH}_2)_2(\text{CF}_2)_{n-2}\text{CF}_3$, and $\text{HO}_2\text{C}(\text{CF}_2)_n\text{CF}_3$ ($n = 2, 4, 6, 8$). (b) Plot of log-current density ($\log|J|$) against the number of methylene (CH_2) and/or difluoromethylene (CF_2) groups at -0.5 V. The values of β for CH_2 and CF_2 are similar; the numbers are inserted in the figure. The intercepts at the y axis indicate the extrapolated log-current densities ($\log|J_0|$) for $\text{HO}_2\text{C}(\text{CH}_2)_n\text{CH}_3$ (■, $\log|J_0| = 2.8 \pm 0.2$), $\text{HO}_2\text{C}(\text{CH}_2)_2(\text{CF}_2)_{n-2}\text{CF}_3$ (■, $\log|J_0| = 1.5 \pm 0.3$), and $\text{HO}_2\text{C}(\text{CF}_2)_n\text{CF}_3$ (□, $\log|J_0| = 1.3 \pm 0.2$), when $n = 0$.

previously studied n -alkanoates.¹⁹ Because the R^{F} and R^{H} SAMs may adopt different molecular structures (e.g., helical $(\text{CF}_2)_n$ versus trans-extended $(\text{CH}_2)_n$ conformations), we do not estimate the width of the tunneling barrier (d in eq 1) from an estimation of the extended length of the molecules in nm, but instead empirically from the number of carbon atoms being considered. The current density across junctions containing fluorinated SAMs decreased exponentially with an increasing length of the fluorinated n -alkyl chain (Figure S1 summarizes histograms of $\log|J|(-0.5 \text{ V})|$ derived from the measurements). None of these junctions rectified current. A linear-least-squares

fitting for each series yielded a slope ($R^2 \geq 0.99$) and an intercept at the y axis.

Values of β for changes in the length of $(\text{CF}_2)_n$ chain ($\beta = (1.15 \pm 0.02)n_{\text{CF}_2}^{-1}$) and corresponding $(\text{CH}_2)_n$ chains ($\beta = (1.05 \pm 0.02)n_{\text{CH}_2}^{-1}$) were apparently slightly different in SAMs of perfluoroalkanoates and n -alkanoates. In the Simmons equation,^{12,13,55,56} the value of β is determined by an effective height of the barrier (ϕ , eV); for hole tunneling,^{2,3} ϕ (assumed to be a simple rectangle) is the difference in energy (eV) between the Fermi level (E_{F}) of the electrode and the HOMO of the molecule. Although the chemical and electronic properties of R^{F} and R^{H} chains are different, we are not able to attribute (with confidence) the small difference (ca. 10%) between the values of β for CF_2 and CH_2 to differences in their frontier orbital energies. It is possible that CF_2 has a lower HOMO energy (by 1 eV) than that of CH_2 ,³⁴ but the work function of R^{F} -bound Ag is also increased by approximately 1 eV (relative to bare Ag).⁴⁸ The simultaneous shifts in the HOMO energy of molecules and the work function of the SAM-bound electrode make ϕ of CF_2 and CH_2 indistinguishable (to the level of granularity that we can detect). A study using inelastic electron tunneling spectroscopy reported a similar conclusion.²⁹

The C–F//Ga₂O₃ Interface Is Responsible for Low Tunneling Currents of R^{F} SAMs. The intercepts (when $n = 0$; Figure 1b) represent the extrapolated value of the log-current density ($\log|J_0|$) for junctions having only the carboxylate anchoring group (A), the distal methyl or trifluoromethyl group (T), and their interfaces with the bottom and top electrodes. We refer to the current density for a junction containing $\text{CH}_3//\text{Ga}_2\text{O}_3$ interface as $J_{0,\text{CH}_3//\text{Ga}_2\text{O}_3}$ ($\log|J_{0,\text{CH}_3//\text{Ga}_2\text{O}_3}| = 2.8 \pm 0.2$ for n -alkanoates) and $\text{CF}_3//\text{Ga}_2\text{O}_3$ as $J_{0,\text{CF}_3//\text{Ga}_2\text{O}_3}$ ($\log|J_{0,\text{CF}_3//\text{Ga}_2\text{O}_3}| = 1.3 \pm 0.2$ for perfluoroalkanoates and $\log|J_{0,\text{CF}_3//\text{Ga}_2\text{O}_3}| = 1.5 \pm 0.3$ for $2\text{H},2\text{H},3\text{H},3\text{H}$ -perfluoroalkanoates; Table 1). The difference in J_0 (up to a factor of ~ 30) between the R^{F} and R^{H} SAMs, and the similarity in β for compounds that interchange CH_2 and CF_2 groups, suggest that the C–F//Ga₂O₃ interface is responsible for the reduction in J_0 . To verify the influence of the C–F//Ga₂O₃ interface on the rates of charge transport, we replaced only the distal CH_3 with a CF_3 at the *para*-position of toluic acid ($\text{HO}_2\text{C}(\text{C}_6\text{H}_4)\text{CH}_3$), 3-tolylpropanoic acid ($\text{HO}_2\text{C}(\text{CH}_2)_2(\text{C}_6\text{H}_4)\text{CH}_3$), and 4'-methyl-biphenyl-4-carboxylic acid ($\text{HO}_2\text{C}(\text{C}_6\text{H}_4)_2\text{CH}_3$). We do not expect this structural change (i.e., the substitution of CH_3 for a

Table 1. Summary of Current Densities (A/cm^2) Derived from EGaIn-Based Junctions Comprising SAMs of $\text{HO}_2\text{C}(\text{CH}_2)_n\text{CH}_3$, $\text{HO}_2\text{C}(\text{CH}_2)_2(\text{CF}_2)_{n-2}\text{CF}_3$, and $\text{HO}_2\text{C}(\text{CF}_2)_n\text{CF}_3$ ($n = 2, 4, 6, 8$) at -0.5 V^a

n	$\text{HO}_2\text{C}(\text{CH}_2)_n\text{CH}_3$ $\log J \pm \sigma_{\log}$	$\text{HO}_2\text{C}(\text{CH}_2)_2(\text{CF}_2)_{n-2}\text{CF}_3$ $\log J \pm \sigma_{\log}$	$\text{HO}_2\text{C}(\text{CF}_2)_n\text{CF}_3$ $\log J \pm \sigma_{\log}$
2	1.90 ± 0.10	0.5 ± 0.1	0.24 ± 0.06
4	1.00 ± 0.20	-0.4 ± 0.2	-0.72 ± 0.07
6	-0.06 ± 0.06	-1.3 ± 0.3	-1.70 ± 0.20
8	-0.80 ± 0.20	-2.4 ± 0.2	-2.75 ± 0.09
0	$\log J_0 = 2.8 \pm 0.20$	$\log J_0 = 1.5 \pm 0.3$	$\log J_0 = 1.30 \pm 0.20$
	$\beta = (1.05 \pm 0.02)n_{\text{CH}_2}^{-1}$	$\beta = (1.11 \pm 0.03)n_{\text{CH}_2/\text{CF}_2}^{-1}$	$\beta = (1.15 \pm 0.02)n_{\text{CF}_2}^{-1}$

^aA linear-least-squares fit for each series of SAMs (Figure 1) yielded a slope and an intercept at the y axis; the former provides the attenuation factor, β , of the repeating units, and the latter represents the value of the $\log|J_0|$ when $n = 0$. We adapted the result of junction measurements of $\text{HO}_2\text{C}(\text{CH}_2)_n\text{H}$ from ref 19.

CF₃ at the *para*-position) to influence significantly the order and packing of the SAM. The C–F bonds (~1.4 Å) are ~30% longer than C–H bonds (~1.1 Å), but we estimate that this difference in length (and a consequent change in the width of the barrier) would have only a small effect on $J(V)$ (by a factor of ~1.3);⁵⁷ the surface energy (hydrophobicity or polarizability) of the SAM, however, may change due to the organization of the C–F bonds.³⁰

The results confirm the hypothesis that the interaction between the distal fluorinated substituents with the Ga₂O₃ influences the shape of the barrier of the T//Ga₂O₃ interface and reduces the rate of charge transport across the junctions. Figure 2 shows histograms of $\log|J(-0.5\text{ V})|$ of

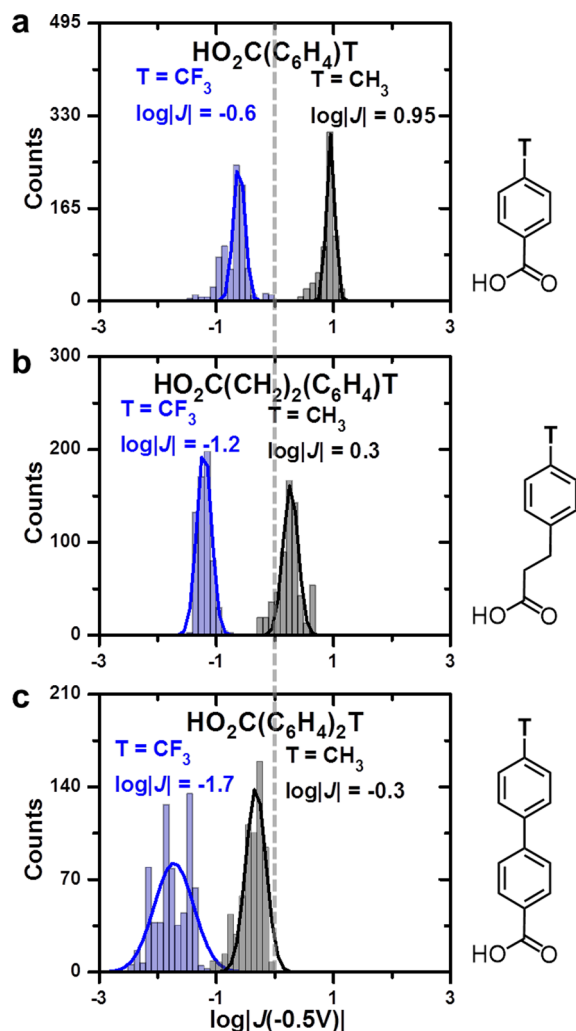


Figure 2. Histograms of $\log|J(-0.5\text{ V})|$ data derived from HO₂C-(CH₂)_n(C₆H₄)_mT, where T = CH₃ (black) and CF₃ (blue): (a) HO₂C(C₆H₄)T, (b) HO₂C(CH₂)₂(C₆H₄)T, and (c) HO₂C(C₆H₄)₂T. The dotted line aligns $J(-0.5\text{ V})$ at 1.0 A/cm².

HO₂C(CH₂)₂(C₆H₄)T and HO₂C(C₆H₄)_mT, where $m = 1, 2$ and T = CH₃ (black traces) or CF₃ (blue traces). These three pairs of compounds show a consistent difference of $\Delta\log|J|_{\text{mean}} \approx 1.5$ between SAMs terminated with T = CH₃ and CF₃; that is, the rates of charge transport across junctions having a CF₃//Ga₂O₃ interface are lower by a factor of ~30 than those across molecularly analogous junctions with a CH₃//Ga₂O₃ interface.

The Nature of the T//Ga₂O₃ Interface Does Not Influence the Attenuation of Current across Other Components in the Junctions. We studied another series of junctions, comprising SAMs of ω -phenyl-alkanoic acids (HO₂C(CH₂)_n(C₆H₄)T, here $m = 0, 2, 4$ and T = H or F; Figure 3a), to determine the influence of the C–F//Ga₂O₃

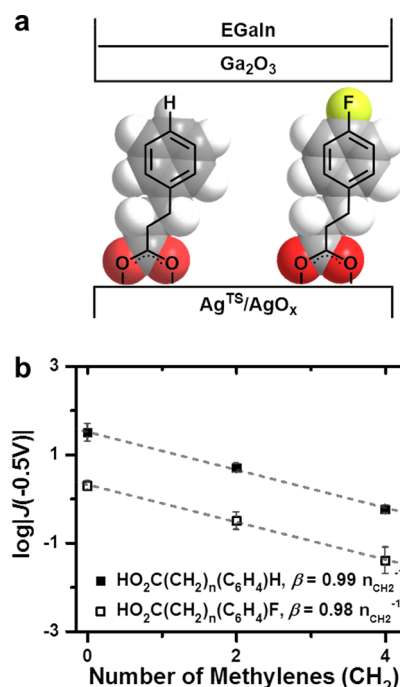


Figure 3. (a) Cartoon representation of the Ag^{TS}O₂C(CH₂)_n(C₆H₄)T //Ga₂O₃/EGaIn junction, where $n = 0, 2, 4$ and T = H or F. (b) A plot of $\log|J(-0.5\text{ V})|$ against the number of methylene (CH₂) groups at -0.5 V . The intercepts at the y axis indicate $\log|J_0|$ for ω -phenyl-alkanoates ($1.5 \pm 0.2\text{ A/cm}^2$) and ω -(4-fluorophenyl)-alkanoates ($0.3 \pm 0.2\text{ A/cm}^2$) when $n = 0$. The separation between the two lines suggests that the presence of fluorine at the SAM//Ga₂O₃ interface reduces the value of J_0 , and the tunneling current, by a factor of 16.

interface on β across the rest of the molecule in the junction (i.e., the methylene groups in the SAMs). Figure 3b shows a plot of $\log|J(-0.5\text{ V})|$ versus the number of CH₂ groups in the structure of HO₂C(CH₂)_n(C₆H₄)T with T = H (■) or F (□). Linear-least-squares fits for these two series give indistinguishable values of β for changes in the length of the polymethylene chain (CH₂)_n ($\beta = (0.98\text{--}0.99)n_{\text{CH}_2}^{-1}$), but distinct intercepts (at the y axis) when $n = 0$. The indistinguishable values of β agree with observations made by us and others in alkane-based systems,^{11,19,58–60} and show that the terminal C–F//Ga₂O₃ interface does not influence the barrier height associated with a polymethylene segment in the interior of the SAM. The difference of the intercepts ($\Delta\log|J(-0.5\text{ V})| = 1.2$) in Figure 3b (and the difference in the values of \log -current densities in Figure 4) also agrees with our observation of low tunneling currents for SAMs terminated with a CF₃ group (Figure 1 and Table 1). The results from these compounds demonstrate that the C–F//Ga₂O₃ interface is part of the tunneling barrier, as expected from prior work,²⁰ and is—in these compounds—responsible for the reduction in J_0 .

We have explored previously the influence of terminal groups T—ranging from aliphatic, simple aromatic, polar, and Lewis-

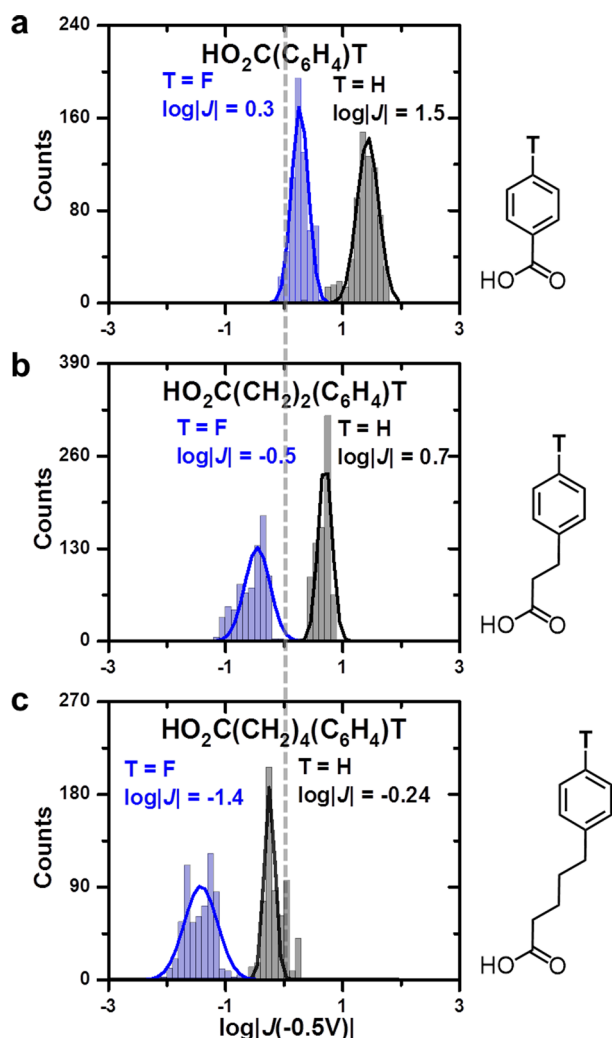


Figure 4. Histograms of $\log|J(-0.5 \text{ V})|$ data derived from $\text{HO}_2\text{C}-(\text{CH}_2)_n(\text{C}_6\text{H}_4)\text{T}$, where T = H (black) and F (blue): (a) $\text{HO}_2\text{C}(\text{C}_6\text{H}_4)\text{T}$, (b) $\text{HO}_2\text{C}(\text{CH}_2)_2(\text{C}_6\text{H}_4)\text{T}$, and (c) $\text{HO}_2\text{C}-(\text{CH}_2)_4(\text{C}_6\text{H}_4)\text{T}$. The dotted line aligns $J(-0.5 \text{ V})$ at $1.0 \text{ A}/\text{cm}^2$.

acidic and -basic functional groups that form van der Waals, hydrogen, and/or ionic interactions at the T// Ga_2O_3 interface—on the shape of the tunneling barrier.^{20–22} A comparison of rates of charge transport to *n*-alkanes (using a standard set of reference compounds) suggested that increasing the strength of the interaction—from a van der Waals interface to a hydrogen-bonded and/or ionically bonded interface—did not significantly influence the height of the barrier at the T// Ga_2O_3 interface. Among these molecules, SAMs of $\text{S}(\text{CH}_2)_4\text{CONH}-(\text{CH}_2)_2\text{CF}_3$ seemed to be the exception; we measured a lower $J(V)$ (by a factor of 3) relative to that of a standard *n*-alkanethiolate of similar length.²¹ We do not entirely understand why this factor is not the factor of 20–30 measured here, but note that the terminal group of $\text{S}(\text{CH}_2)_4\text{CONH}-(\text{CH}_2)_2\text{CF}_3$ in contact with the $\text{Ga}_2\text{O}_3/\text{EGaIn}$ electrode might not be exclusively trifluoromethyl (CF_3). Our recent study of odd/even effects²⁵ suggested that the structure of the terminal group in contact with the Ga_2O_3 might be closer to trifluoroethyl ($-\text{CH}_2\text{CF}_3$) than to CF_3 , due to changes in the supramolecular structure of the SAM; this difference could introduce a C–H// Ga_2O_3 contact at the interface. This present study of perfluorinated SAMs ($\text{O}_2\text{C}(\text{CF}_2)_n\text{CF}_3$), in the context

of the present results, excludes the possibility of a mixed C–F(H)// Ga_2O_3 interface.

CONCLUSIONS

The key experimental result of this study is its demonstration that changing a terminal methyl (CH_3) group to a trifluoromethyl (CF_3) group decreases the rate of tunneling by a factor of 20–30, but changing an internal methylene (CH_2) to a difluoromethylene (CF_2) has only a small effect. It leads to the interesting suggestion that although it may not be possible to increase the tunneling current by lowering the effective height of the barrier at the T// Ga_2O_3 interface by increasing the strength of the interaction between the group T and Ga_2O_3 , it may be possible to decrease the rate of tunneling by raising the height of that part of the barrier by decreasing the strength of this interaction.

We have compared tunneling currents of partially and completely fluorinated *n*-alkanoates and the homologous *n*-alkanoates using EGaIn-based junctions. We conclude the following: (i) The influence of CH_2 and CF_2 groups on the shape of the tunneling barrier is similar (although marginally distinguishable by these measurements). This observation suggests that fluorination in the methylene groups of *n*-alkanes has only a small effect (at the scale that we can detect) on the apparent height (ϕ) (as approximated by eq 1) of the tunneling barrier, perhaps because both the HOMO energy of R^{F} SAMs and the work function of R^{F} -bound Ag are shifted, in ways that partially cancel. (ii) By contrast, fluorination of the terminal group has a large ($\times 20$ – 30) effect and decreases tunneling currents. We conclude that the C–F// Ga_2O_3 interface is the part of the tunneling barrier that is responsible for the reduction of rates of charge transport across R^{F} SAMs.

The mechanism(s) of reduction of tunneling transport across the R^{F} // $\text{Ga}_2\text{O}_3/\text{EGaIn}$ interface (relative to a structurally similar R^{H} // $\text{Ga}_2\text{O}_3/\text{EGaIn}$ interface) is still not uniquely defined by the available data. We suggest four plausible mechanisms, of which (i) and (ii) are less plausible, and (iii) and (iv) are difficult to differentiate.

(i). *Change in Work Function.* C–F bonds or fluorocarbons do not have the highest dipole moment among polar groups (including cyanide and nitro groups) in our prior study,²¹ but these polar terminal groups in alkyl-based SAMs do not significantly influence tunneling currents. Thus, a high dipole moment (and perhaps, in consequence, a high surface potential) does not appear to be the origin of the decrease in tunneling current density observed on converting a terminal CH_3 group to a CF_3 group.

(ii). *Relative Size of Fluorocarbons and Hydrocarbons.* Although the C–F bonds ($\sim 1.4 \text{ \AA}$) are considerably longer than C–H bonds ($\sim 1.1 \text{ \AA}$), we estimated that this difference in length (and thus in the width of the barrier) would only have a small effect on $J(V)$ (a factor of ~ 1.3). Thus, the greater “size” of terminal fluorinated groups (relative to a CH_3 group) does not seem to be responsible to the decreased rates of tunneling for T = CF_3 .

(iii). *Weak Electronic Interactions at the Interface.* Fluorine has the lowest polarizability ($0.56 \times 10^{-24} \text{ cm}^3$) of all elements,³¹ and this low polarizability might decrease the strength of the van der Waals interactions across the interface, decrease the magnitude of electronic coupling between the R^{F} SAMs and the top electrodes, and therefore, reduce the rates of tunneling across the interface.

(iv). *Poor Wetting at the Interface.* Fluorocarbons are both hydrophobic and lipophobic, but the Ga₂O₃ layer of the top EGaIn electrode is hydrophilic.⁵⁴ This difference in surface energy could lead to weak physical contact (wetting) of the C–F surface by the flexible Ga₂O₃ surface membrane at the C–F//Ga₂O₃ interface, and result in a lower area of effective electrical contact for a F//Ga₂O₃ interface than for a H//Ga₂O₃ interface.¹¹

■ ASSOCIATED CONTENT

■ Supporting Information

Detailed experimental procedure, histograms of current densities, and summary of junction measurements. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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Notes

The authors declare no competing financial interest.

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