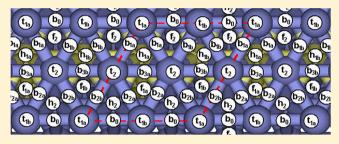
# Mechanism for Oxygen Reduction Reaction on Pt<sub>3</sub>Ni Alloy Fuel Cell **Cathode**

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ABSTRACT: We use quantum mechanics, density functional theory at the PBE level, to predict the binding-site preferences and reaction barriers for all intermediates involved in the oxygen reduction reaction (ORR) on the low energy surface of Pt<sub>3</sub>Ni alloy. Here we calculate that the surface layer is Ni depleted (100% Pt) while the second layer is Ni enriched (50% Pt) as shown by experiment. Even though the top layer is pure Pt, we find that the sublayer Ni imposes strong preferences in binding sites for most intermediates, which in turn strongly influences the reaction barriers. This strong



preference leads to a strong site dependence of the barriers. Considering water as the solvent, we predict that, at low coverage of O<sub>ad</sub> and OH<sub>ad</sub>, the barrier for the rate-determining step is 0.81 eV, whereas, at high coverage, this barrier decreases to 0.43 eV. It can be compared to a barrier of 0.50 eV for pure Pt, explaining the improved ORR rate for the Pt<sub>3</sub>Ni alloy. We report the results both for gas phase and for aqueous phase environments.

# 1. INTRODUCTION

The efficiency of the oxygen reduction reaction (ORR), 4H<sup>+</sup> +  $4e- + O_2 \rightarrow 2H_2O$ , at the cathode of a polymer electrolyte membrane fuel cell (PEMFC) is a critical issue for commercial application of this type of fuel cells. 1-4 The best current catalysts are Pt and Pt-based binary alloys, such as Pt<sub>3</sub>Ni.<sup>5,6</sup> The origin of the superior performance of the Pt<sub>3</sub>Ni alloy has not been clearly understood yet. Some researchers believe it is due to the shift of the d-band center to the desired region that occurs due to alloying Pt with Ni. 7,8 Others came to the conclusion that alloying makes  $OH_{ad}$  removal favorable, increasing the surface area available for O<sub>2</sub> binding.<sup>5</sup> It has also been argued that alloying Pt with Ni or Co decreases the surface lattice parameters to values optimal for ORR.9 A prominent property of Pt<sub>3</sub>Ni and Pt<sub>3</sub>Co is the strong surface segregation observed in experiments.  $^{10-12}$  Our quantum mechanics (QM) calculations, density functional theory (DFT) at the Perdew-Becke-Ernzerhof (PBE) level, using a twodimensional slab model find the similar segregation effect for the Pt<sub>3</sub>Ni alloy which results in the surface structure with 100% Pt in the first layer, 50% Pt in the second layer, and 75% Pt in deeper layers.<sup>13</sup> This strong segregation to form a pure Pt surface layer (similar to core-shell systems where the surface is also pure Pt) is supposed to be important in ensuring the improved ORR activity of these alloy catalysts. A recent study of Matanovic et al. 14 argues that the sublayer concentration directly influences the overpotential. However, to our knowledge, no papers have been published that explore the influence of the atomic level configuration for alloying atoms to the reaction mechanism and barriers simultaneously by taking into account solvent effects.

In our study, we used QM calculations to study the unique binding-site preferences due to the placement of sublayer

alloying atoms for all intermediates involved in the ORR on the segregated surface of Pt<sub>3</sub>Ni and the consequent changes to the reaction barriers and mechanisms. We propose a new detailed atomistic level chemical mechanism explaining the increased reactivity of the Pt<sub>3</sub>Ni alloy. In particular, we show that subsurface Ni has a strong influence on the binding energies and induces a coverage dependence for the preferred ORR mechanism.

#### 2. METHODOLOGY

The Pt<sub>3</sub>Ni(111) alloy surface was modeled as a two-dimensionally infinite periodic slab with four atoms per cell and six layers of atoms. We consider the atomic Pt composition as 100-50-75-75-75-75% Pt, as observed experimentally 10 and calculated theoretically. 13,15,16

All calculations used the PBE functional of DFT. We applied small core norm-conserving angular momentum projected pseudopotentials<sup>17–20</sup> in which the 3p, 3d, and 4s electrons of Ni and the 5p, 5d, and 6s electrons on the Pt are treated explicitly with 16 electrons for neutral Ni and Pt. The SeqQuest software<sup>21</sup> with optimized double-ζ plus polarization quality Gaussian-type orbitals on Pt and Ni was employed for our calculations. The periodic cell parameter of the slab corresponds to that of the optimized Pt<sub>3</sub>Ni bulk structure and 0.08% smaller than the experimental value.<sup>22</sup> All charges came from the Mulliken population analysis of the DFT wave function.

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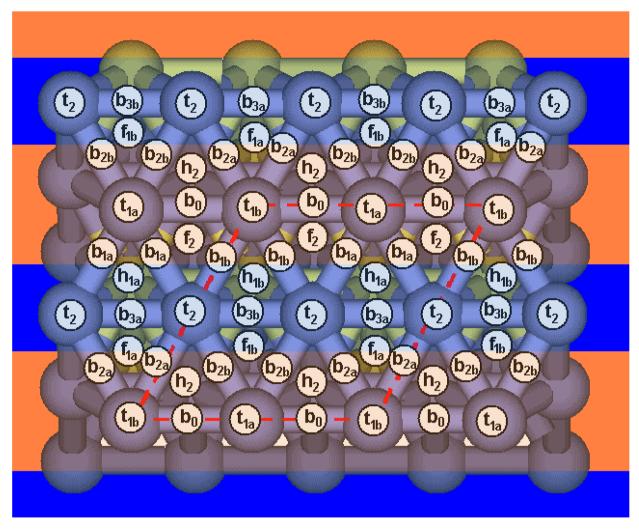


Figure 1. Binding sites on the  $Pt_3Ni(111)$  surface. The blue and orange stripes indicate the partitioning of the Pt surface into two regions induced by the sublayer Ni. The  $O_{22}$  O, H, OH, and OOH species prefer to move only within the orange stripes.

To represent the effects of solvent stabilization, we use the implicit model developed earlier.<sup>23</sup> The Mulliken charges were used as inputs to our implicit continuum solvation model based on the Poisson-Boltzmann equation. We showed that this implicit solvation model<sup>23</sup> correctly reproduces the solvation energies of O<sub>ad</sub>, H<sub>ad</sub>, OH<sub>ad</sub>, and H<sub>2</sub>O<sub>ad</sub> calculated using explicit water layers. Implicit solvation can be applied both with periodic boundary conditions and for finite systems. Using explicit water requires thousands of water molecules to be treated dynamically, which is not yet practical with current QM methods. Solvation greatly affects the kinetics of the ORR on Pt and Pd surfaces. Theoretically, without solvation, Pd would outperform Pt as the ORR catalyst.<sup>23</sup> We use the same approach in this study and expect that charge transfer between Pt and Ni atoms may induce even larger dipoles and hence more significant solvent effects.

The barriers were calculated applying the nudged elastic band (NEB) method.<sup>24,25</sup> The energy, required to move a surface species (from a remote unit cell) to the starting site, is considered in deciding the preferred mechanism so that the site energy would not dominate the barrier. However, this site energy is not included in the barrier, because otherwise the barriers would correspond to the barriers for extremely low coverage. The reactants are assumed to be inside the same unit cell.

#### 3. RESULTS AND DISCUSSION

**3.1. Binding of ORR Intermediates.** *3.1.1. Binding Site Notation.* First, we studied the preference of H, O, OH,  $H_2O$ ,  $O_2$ , and OOH on various binding sites shown in Figure 1. Generally, a closest packed (111) surface of FCC structured metals has four types of sites:

- on-top, bonded to one Pt  $(\mu_1)$ , denoted as t
- bridging, between two Pt  $(\mu_2)$ , denoted as b
- bridging, between three Pt  $(\mu_3)$  but in the FCC position (not above atoms of the top or second layer), denoted as f
- bridging, between three Pt  $(\mu_3)$  but in the HCP position (above atoms of the second layer), denoted as h

However, due to strong segregation, the  $Pt_3Ni$  surface has 100% Pt in the first layer, 50% Ni and 50% Pt in the second layer, and 25% Ni and 75% Pt in the four remaining layers. We find that the binding energies of the intermediates to the pure Pt layer strongly depend on the nature of the second layer atoms. The various cases are tabulated in Figure 1. Figure 2 shows details of the difference between sites.

For the first and second layers, there are two types of top sites:  $t_1$  with one Ni neighbor in the second layer and  $t_2$  with two Ni neighbors. Considering also the third layer, we can distinguish  $t_{1a}$  with no Ni in the third layer directly beneath

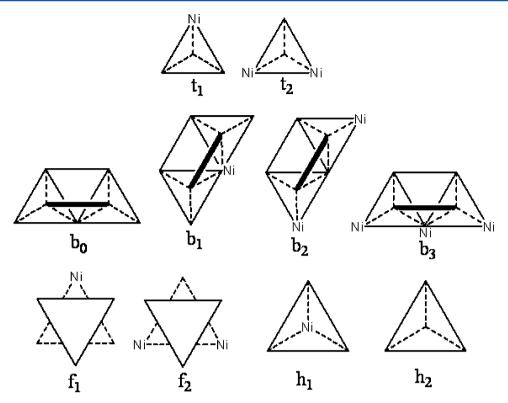


Figure 2. Illustration of various binding sites on the  $Pt_3Ni(111)$  surface. For top sites  $t_1$  and  $t_2$ , the triangle indicates the sublayer atoms.  $t_1$  has one Ni atom beneath it, while  $t_2$  has two. For bridge sites, the bridge itself is shown as the thick black line, while the two termini of the black line connect the two surface atoms forming the bridge site. The trapezoids beneath are sublayer atoms.  $b_0-b_3$  have from 0 to 3 Ni atoms in the sublayer. An FCC site is in the center of a surface triangle (shown as a solid triangle).  $f_1$  and  $f_2$  differ in the sublayer triangle beneath the surface triangle.  $f_1$  has one Ni beneath it, while  $f_2$  has two. HCP sites are also in the center of a surface triangle. They have one sublayer atom beneath it. For  $h_1$ , it is Ni, while, for  $h_2$ , it is Pt.

the surface and  $t_{1b}$  with one Ni. All  $t_2$  sites are the same (see Figure 2).

For the top two layers, there are four  $\mu_2$  bridge sites, depending on the number of Ni atoms underneath it:  $b_0$ ,  $b_1$ ,  $b_2$ , and  $b_3$  with 0, 1, 2, and 3 Ni atoms in the second layer. With adding the third layer, there appear to be two subtypes for  $b_1$ ,  $b_2$ , and  $b_3$ , depending on the distance to the Ni atom in the third layer. We denote the subtypes closer to the third layer Ni as  $b_{1a}$ ,  $b_{2a}$ , and  $b_{3a}$  and the others as  $b_{1b}$ ,  $b_{2b}$ , and  $b_{3b}$ , respectively (Figure 2).

Considering only two top layers, two FCC sites can be distinguished:  $f_1$  and  $f_2$  with one and two Ni atoms in the sublayer triangle, but adding the third layer splits  $f_1$  into  $f_{1a}$  and  $f_{1b}$  with  $f_{1a}$  on top of the third layer Ni and  $f_{1b}$  on top of the third layer Pt.

Similarly, there are two HCP sites related to the two top layers:  $h_1$  and  $h_2$ . Here,  $h_1$  is on top of the Ni sublayer, while  $h_2$  is on top of the Pt sublayer. Adding the third layer splits the  $h_1$  site into  $h_{1a}$  and  $h_{1b}$  with one Ni atom and without Ni atoms in the projected triangle of the third layer atoms, as shown in Figure 2.

All the above sites can be partitioned into two categories: the sites closer to the sublayer Pt atoms (the orange region in Figure 1) and the sites closer to the sublayer Ni atoms (the blue region in Figure 1). As we will show later, the orange region is universally preferred.

The binding energies (BE) are calculated as the energy gain for species to adsorb to the surface, i.e.,

$$BE_{gas} = E_{surf+M,gas} - E_{surf,gas} - E_{M,gas}$$

where M is an adsorbate. For the solution phase, the solvent stabilization is added directly to the binding energy in order to isolate the influence of water. This leads to

$$BE_{solv} = E_{surf+M,solv} - E_{surf,solv} - E_{M,gas}$$

The solvent phase binding energy does not include the solvent effect of the adsorbate itself, because most of the species are radicals that do not have well-defined solvation energies for comparison. The lack of these energy data for radical species does not affect the barrier calculations, since the reactants, products, and transition states are all surface species and the differences in individual solvation energies eventually cancel out.

All barriers are calculated as the energy difference between a transition state and surface reactants. We disregard the energy that reactants need to migrate from the globally preferred sites to the reacting sites, as this barrier would correspond only to very low coverage. This condition is not typical for conventional fuel cell operation.

3.1.2. H Binding. The preferred binding site for H on the  $Pt_3Ni$  surface is  $t_{1a}$  with a binding energy of 2.70 eV, followed by  $t_{1b}$  and  $b_0$  with a binding energy of 2.60 eV in the gas phase.

On the other hand, the  $b_{3a}$ ,  $b_{3b}$ , and  $t_2$  sites in the orange region have binding energies of 2.39–2.44 eV in the gas phase, 0.26–0.31 eV weaker than the binding energy of the preferred binding site. The barrier for  $H_{ad}$  migration within the orange region ( $b_0 \rightarrow t_{1a} \rightarrow b_0 \rightarrow t_{1b} \rightarrow b_0$ ) is approximately 0.10 eV, while the barrier for  $H_{ad}$  to migrate between adjacent orange regions (through the blue region) is 0.26 eV. Hence, hydrogen prefers to stay within the orange region. We will show below

that this preference is universal for all surface species. For clarity, the orange region is denoted as "the preferred region".

Under solvation, the preferred sites become  $t_{1a}$ ,  $b_{1a}$ , and  $f_{1b}$ , with binding energies of 2.78–2.83 eV, followed by  $t_{1b}$ ,  $b_{0}$ ,  $b_{1ab}$ ,  $t_{2}$ ,  $t_{1a}$ ,  $t_{1ab}$ , and  $t_{2}$  with binding energies of 2.68–2.74 eV.

The  $b_{3a}$ ,  $b_{3b}$ , and  $t_2$  sites in the orange region have a binding energy of 2.48–2.54 eV under solvation, which is weaker by 0.3 eV than that of the preferred binding site.

Comparing with the gas phase results, we conclude that solvation does not alternate the preferred region (orange region). The barrier for the  $H_{\rm ad}$  migration within the orange region is as low as 0.08 eV, while the barrier for  $H_{\rm ad}$  to migrate between adjacent orange regions is 0.30 eV. The difference of 0.22 eV is higher than the corresponding difference of 0.16 eV for the gas phase. Therefore, solvation makes migration between adjacent orange regions more difficult. Consequently, the blue region serves as a barrier region preventing hydrogen from diffusing between different orange stripes.

For the pure Pt surface, the binding energy of H is 2.70-2.80 eV in the gas phase and 2.81-2.81 eV in the solvated phase. There are no "forbidden" regions in this case. This allows  $H_{ad}$  to migrate easily in all directions to react with other species.

3.1.3. O Binding. On pure Pt,  $O_{ad}$  binds strongly to the FCC site with a net energy of 3.66 eV in the gas phase and 4.36 eV in the solvated phase. The huge solvation stabilization arises from electrostatics due to appearance of the strong dipole at surface  $O_{ad}$  atoms.

For  $Pt_3Ni$ , the binding energy for  $O_{ad}$  depends dramatically on the site. The  $f_2$  site is preferable in the gas phase with a binding energy of 3.47 eV, followed by the  $f_{1a}$  site with a binding energy of 3.17 eV. The next best sites in the gas phase are  $b_0$ ,  $f_{1b}$ , and  $h_2$ , which are weaker than  $f_2$  by 0.40 eV.

With solvent,  $f_2$  and  $f_{1b}$  become dominant with binding energies of 4.49 and 4.28 eV, respectively. All other binding sites are at least 0.60 eV less stable than  $f_2$  and have small barriers for O to migrate into the  $f_2$  site, except for the case with extremely high coverage. The strong stabilization of the  $f_2$  site is due to two electropositive Ni with the Mulliken charge of 0.14–0.18e in the second layer that enhance the charge transfer to the surface  $O_{ad}$ , stabilizing it. This is illustrated by the surface total charge of 0.079 on  $f_1$  and 0.114 on  $f_{2a}$  and  $f_{2b}$ .

The binding energy for the FCC site on the pure Pt surface is 0.19 eV stronger than that for the  $f_2$  site in the gas phase but 0.13 eV weaker with solvation.

The smaller Ni atoms allow the surface Pt above two Ni to contract the distance by 0.29 Å toward the bulk values, whereas the corresponding change for the surface Pt above one Ni is 0.19 Å.

These results indicate that  $O_{ad}$ , formed from  $O_2$  dissociation, strongly prefers to occupy the  $f_2$  site and no further migration occurs to the other sites both in the gas phase and solution.

3.1.4. OH Binding. On pure Pt, OH has almost the same binding energy at all sites, 2.22–2.28 eV in the gas phase and 2.57–2.77 eV in solution.

For  $Pt_3Ni$ , the best site in the gas phase is  $b_0$  with a binding energy of 2.42 eV. Here, the OH-bond tilts toward  $h_{1b}$  (away from  $f_2$ ). The  $t_{1a}$ ,  $t_{1b}$ ,  $f_2$ ,  $h_2$ , and  $b_{1a}$  sites are less stable than  $b_0$  by 0.10–0.20 eV. The barrier for OH to migrate across the preferred regions is 0.29 eV, while the barrier for OH migration within the same preferred region is negligible. Hence, similar to the H binding, OH selectively binds and stays within the preferred region.

With solvation,  $t_{1a}$  is the most preferred site with a binding energy of 2.83 eV, followed by  $f_2$ ,  $b_0$ , and  $t_{1b}$  with a binding

energy of 2.73–2.75 eV. For comparison, OH binds much weaker to  $t_2$ ,  $b_{3a}$ , and  $b_{3b}$  with a binding energy of 2.24–2.47 eV. Thus, the preferred region remains the same as in the gas phase (orange region in Figure 1). The in-region OH<sub>ad</sub> migration barrier is 0.10 eV, and the across-region barrier is 0.36 eV (estimated by the binding energy of  $b_{3a}$  and  $b_{3b}$ ).

At the other binding sites,  $OH_{ad}$  has a strong preference to migrate to the preferred region. This differs dramatically from the case of the pure Pt surface where the binding energy of OH at all possible sites ranges within 0.06 eV in the gas phase and 0.20 eV in the solvated phase, indicating the ability for easy OH migration.

3.1.5.  $O_2$  Binding. For pure Pt, we find that the binding energy of  $O_2$  is 0.46 eV in the gas phase and 0.87 eV in the solvated phase, with the difference for various sites ranging within 0.11 and 0.17 eV for gas phase and solution, respectively.

For the  $Pt_3Ni$  surface,  $O_2$  binds most strongly to  $b_{0}$ ,  $b_{1a}$ ,  $b_{1b}$ , and  $f_2$  in the gas phase. The corresponding binding mmaenergies vary from 0.33 to 0.59 eV in the gas phase and from 0.62 to 0.76 eV in solution. All other sites are less favorable by at least 0.3 eV. Thus,  $O_2$  has a small preferred binding region near the  $f_2$  site on  $Pt_3Ni$  which is adjacent to the  $f_2$  site for  $O_{ad}$  and to the  $b_0$  site for  $OH_{ad}$ . This small region is part of the preferred region (orange region), and the barrier to migrate between preferred regions is at least 0.32 eV in the gas phase and 0.38 eV in the solvated phase (estimated by the binding energy of  $b_{3a}$  and  $b_{3b}$ ).

Given that H also prefers this region and would likely be nearby, this implies a favorable migration pathway for OOH formation.

3.1.6. OOH Binding. For pure Pt, OOH binds to the top sites with the terminal O bonded to the Pt and the OOH plane parallel to the surface. OOH prefers to have the O-O bond heading to an adjacent Pt atom. This leads to a binding energy of 1.06 eV in the gas phase and 1.52 eV in solution.

For  $Pt_3Ni$ , we find that  $t_{1a}$  is quite favorable with the binding energy of 1.05 eV in the gas phase and 1.55 eV in the solvated phase. Here the preferred orientation for the O-O bond is also toward the adjacent Pt atom, similar to pure Pt.

All other sites are not reachable for OOH, leading instead to dissociation. Thus, once formed, OOH cannot migrate on the surface.

3.1.7.  $H_2O$  Binding.  $H_2O$  binds only to top sites with binding energies of 0.18–0.20 eV in the gas phase and 0.57–0.60 eV in solution, all close to the corresponding values of 0.22 and 0.58 eV for pure Pt. Comparing with the 0.40 eV solvent stabilization of bulk  $H_2O$ , the surface  $H_2O$  shows positive binding in both cases.

Since  $H_2O$  does not bind to bridge, FCC, and HCP sites, migration of  $H_2O_{ad}$  from one top site to another is through adsorption and dissociation. The migration barrier is 0.20 eV both in the gas phase and solution (the 0.60 eV binding energy minus the 0.40 eV solvation of  $H_2O_{ad}$ ,  $H_2O$  is considered always solvated).

3.1.8.  $H_2O_2$  Binding.  $H_2O_2$  has a binding energy of 0.23–0.31 eV on the Pt<sub>3</sub>Ni surface in the gas phase and 0.58–0.67 eV in the solvated phase, close to the 0.27 and 0.61 eV for pure Pt. Similar to the  $H_2O$  case,  $H_2O_2$  migrates through desorption and readsorption.

3.1.9. Summary of Binding. Summarizing, we find (see Tables 1 and 2) that

- O strongly prefers f<sub>2</sub>.
- $O_2$  can occupy the  $b_0$ ,  $b_{1a}$ ,  $b_{1b}$ , and  $f_2$  sites.
- H prefers to move within in the orange region, allowing it to attack O at the f<sub>2</sub> site to form OH at the b<sub>0</sub>, t<sub>1a</sub>, or t<sub>1b</sub> sites.

Table 1. Binding Energies (eV) of ORR Intermediates at Different Sites on Pt<sub>3</sub>Ni and Pt in the Gas Phase

BE	BE	Н	O	ОН	$O_2$	ООН	$H_2O$	$H_2O_2$
Pt <sub>3</sub> Ni	t <sub>1a</sub>	-2.70	-2.48	-2.29		$-1.05^{a}$	-0.20	
	$t_{1\mathrm{b}}$	-2.60	-2.37	-2.22		$-0.97^{a}$	-0.18	
	$t_2$	-2.44	-2.16	-2.00		$-0.98^{a}$	-0.18	
	$b_0$	-2.60	-3.09	-2.42	-0.59			-0.23
	$b_{1a}$	-2.57	-2.98	-2.25	-0.47			-0.29
	$b_{1b}$	-2.48	-2.95	-2.10	-0.39			-0.29
	$b_{3a}$	-2.39	-2.67	-1.96	-0.26			-0.29
	$b_{3b}$	-2.39	-2.62	-1.85	-0.24			-0.29
	$b_{2a}$	-2.46	-2.76	-1.98	-0.27			-0.31
	$b_{2b}$	-2.52	-2.75	-2.01	-0.25			-0.31
	$f_2$	-2.55	-3.47	-2.29	-0.33			
	$f_{1a}$	-2.45	-3.17	-1.92	0.04			
	$f_{1b}$	-2.49	-3.08	-1.89	0.00			
	$h_{1a}$	-2.48	-2.96	-1.90	-0.18			
	$h_{1b}$	-2.47	-2.94	-1.87	-0.09			
	$h_2$	-2.53	-3.06	-2.32	-0.15			
	best	-2.70	-3.47	-2.42	-0.59	-1.05	-0.20	-0.31
Pt	t	-2.80	-2.50	-2.23		-1.06	-0.22	
	ь	-2.70	-3.10	-2.25	-0.40			-0.27
	f	-2.72	-3.66	-2.22	-0.46			
	h	-2.70	-3.28	-2.28	-0.35			
	best	-2.80	-3.66	-2.28	-0.46	-1.06	-0.22	-0.27

<sup>&</sup>quot;OOH can have two orientations: the O-O bond points toward an adjacent Pt atom (preferred, binding energy shown) or toward an FCC/HCP site (binding energies 0.93, 0.77, and 0.78 eV, respectively).

Table 2. Binding Energies (eV) of ORR Intermediates at Different Sites on Pt<sub>3</sub>Ni and Pt in Solution

BE	sites	Н	0	ОН	$O_2$	ООН	$H_2O$	$H_2O_2$
Pt₃Ni	$t_{1a}$	-2.78	-3.10	-2.83		$-1.55^{a}$	-0.61	
	$t_{1b}$	-2.70	-2.95	-2.73		$-1.46^{a}$	-0.57	
	$t_2$	-2.48	-2.66	-2.47		$-1.45^{a}$	-0.58	
	$b_0$	-2.70	-3.70	-2.74	-0.75			-0.58
	$b_{1a}$	-2.83	-3.82	-2.64	-0.66			-0.63
	$b_{1b}$	-2.69	-3.79	-2.55	-0.62			-0.64
	$b_{3a}$	-2.48	-3.25	-2.30	-0.36			-0.6
	$b_{3b}$	-2.54	-3.50	-2.24	-0.37			-0.6
	$b_{2a}$	-2.57	-3.35	-2.32	-0.39			-0.6
	$b_{2b}$	-2.63	-3.38	-2.33	-0.40			-0.6
	$f_2$	-2.74	-4.49	-2.75	-0.76			
	$f_{1a}$	-2.55	-3.80	-2.27	-0.20			
	$f_{1b}$	-2.77	-4.28	-2.28	-0.34			
	$h_{1a}$	-2.69	-3.69	-2.25	-0.38			
	$h_{1b}$	-2.74	-3.89	-2.31	-0.30			
	$h_2$	-2.68	-3.66	-2.66	-0.38			
	best	-2.83	-4.49	-2.83	-0.76	-1.55	-0.61	-0.6
Pt	t	-2.87	-3.09	-2.77		-1.52	-0.58	
	Ь	-2.82	-3.73	-2.63	-0.73			-0.6
	f	-2.85	-4.36	-2.57	-0.87			
	h	-2.81	-3.92	-2.64	-0.70			
	best	-2.87	-4.36	-2.77	-0.87	-1.52	-0.58	-0.6

<sup>&</sup>lt;sup>a</sup>Same notes as for Table 1 except that the binding energies are 1.42, 1.24, and 1.32 eV.

- OH can also move through b<sub>0</sub>, t<sub>1a</sub>, t<sub>1b</sub>, f<sub>2</sub>, and h<sub>2</sub>.
- OOH occupies top sites with the second O orienting nearby Pt.
- H<sub>2</sub>O prefers all top sites, while H<sub>2</sub>O<sub>2</sub> does bridge sites.

These imply a strong regional preference for  $O_2$ , O, H, OH, and OOH to stay in the orange stripes. Of the top sites,  $t_{1a}$  and  $t_{1b}$  are most preferable for all species. Similarly,  $b_0$  is predominantly preferred among bridge sites. Among the 3-fold FCC, HCP,  $f_{1a}$ ,

 $f_{1b}$ ,  $f_2$ ,  $h_{1a}$ ,  $h_{1b}$ , and  $h_2$  binding sites,  $f_2$  is most favorable. Thus, at lower coverage (quarter layer), the  $b_0$ ,  $f_2$ , and  $t_{1a}$  (also  $t_{1b}$ ) sites are preferred, when the surface is exposed to all the intermediates. Only at higher coverage, the adsorbates bind to other sites.

Therefore, the chemistry prefers the orange region. The direct consequence of this is that reaction barriers depend on whether the starting and ending sites are inside or outside of the preferred region. This conclusion also agrees with the experimental result obtained by Markovic et al.<sup>5</sup> on Pt<sub>3</sub>Ni(111), where the fractional coverages of the underpotentially deposited hydrogen and adsorbed hydroxyl species were dramatically decreased by 50% as compared to Pt(111). The blue region has much lower binding energy and hence is much lower covered with H<sub>2d</sub> and OH<sub>2d</sub>.

**3.2. Reaction Barriers and Possible ORR Mechanisms.** Our previous studies<sup>13,23,26,27</sup> showed that six fundamental steps can be involved in the ORR mechanism:

(a)  $O_2$  dissociation:  $O_{2ad} \rightarrow O_{ad} + O_{ad}$ 

(b) OH formation:  $O_{ad} + H_{ad} \rightarrow OH_{ad}$ 

(c)  $H_2O$  formation:  $OH_{ad} + H_{ad} \rightarrow H_2O_{ad}$ 

(d) OOH formation:  $O_{2ad} + H_{ad} \rightarrow HOO_{ad}$ 

(e) OOH dissociation:  $HOO_{ad} \rightarrow OH_{ad} + O_{ad}$ 

(f) Hydration:  $O_{ad} + H_2O_{ad} \rightarrow OH_{ad} + O_{ad}$ 

By including these fundamental steps in an overall mechanism, we distinguish three chemical processes:

O-O bond activation, which can occur via two mechanisms: (1)  $O_2$  dissociation (a) and (2) OOH formation (d) followed by OOH dissociation (e).

OH formation proceeds via two mechanisms as well: OH formation (b) and O hydration (f).

OH consumption. There is only one mechanism,  $H_2O$  formation (c), for this process.

A good catalyst must provide low barriers for all three of these processes and for pathways connecting them.

Starting from the preferred sites, we calculated the barriers for all six steps on Pt<sub>3</sub>Ni in the gas phase and solution. These barriers and their comparison with those for pure Pt are shown in Tables 3 and 4.

Table 3. Reaction Barriers (eV) for ORR Steps on Pt<sub>3</sub>Ni and Pt Surfaces in the Gas Phase

Pt	Pt <sub>3</sub> Ni	high coverage
0.00	0.05	
0.58	1.02	0.92
0.72	0.57	0.67
0.11	0.11	0.36
0.28	0.32	
0.14	0.04	
0.18	0.20	
0.29	0.34	0.00
	0.00 0.58 0.72 0.11 0.28 0.14 0.18	0.00 0.05 0.58 1.02 0.72 0.57 0.11 0.11 0.28 0.32 0.14 0.04 0.18 0.20

"The O hydration on Pt is calculated using a  $3\times3$  cell, because a  $2\times2$  cell is not large enough. For Pt<sub>3</sub>Ni, we used  $4\times2$  and  $2\times4$  cells to keep the correct periodic condition.

Table 4. Reaction Barriers (eV) for ORR Steps on Pt<sub>3</sub>Ni and Pt Surfaces in Solution

reaction barriers	Pt	Pt <sub>3</sub> Ni	high coverage
H <sub>2</sub> dissociation	0.00	0.01	
O <sub>2</sub> dissociation	0.00	0.00	0.24
OH formation	1.09	1.02	0.80
H <sub>2</sub> O formation	0.17	0.25	0.43
OOH formation	0.19	0.17	
OOH dissociation	0.00	0.00	
H-OOH dissociation	0.04	0.22	
O hydration <sup>a</sup>	0.50	0.81	0.44
<sup>a</sup> Same notes as for Table 3.			

3.2.1. Gas Phase Barriers for Pt₃Ni at Low Coverage. O−O Bond Activation. OOH formation with a barrier of 0.32 eV is followed by OOH dissociation with a very small barrier of

0.04 eV, whereas the barrier for the direct dissociation is 1.02 eV. This is similar to the case of pure Pt where OOH formation has a barrier of 0.28 eV compared to 0.58 eV for direct dissociation.

OH Formation. O hydration has a barrier of 0.34 eV, compared to 0.57 eV for the direct OH formation. This preference for the O hydration is similar to the case of pure Pt where the O hydration barrier is 0.29 eV compared to 0.72 eV for the direct OH formation.

OH Consumption.  $H_2O$  formation has a small barrier of 0.11 eV, exactly the same as for the pure Pt case.

Summarizing these three steps, we have the following preferred mechanism (denoted as OOH-form-hydr-gas) for both Pt<sub>2</sub>Ni and Pt.

$$O_{2ad} + H_{ad} \rightarrow OOH_{ad}$$
 $(E_a = 0.32 \text{ eV for Pt}_3\text{Ni and } 0.28 \text{ eV for Pt})$ 
 $OOH_{ad} \rightarrow O_{ad} + OH_{ad}$ 
 $(E_a = 0.04 \text{ eV for Pt}_3\text{Ni and } 0.14 \text{ eV for Pt})$ 
 $O_{ad} + H_2O_{ad} \rightarrow 2OH_{ad}$ 
 $(E_a = 0.34 \text{ eV for Pt}_3\text{Ni and } 0.29 \text{ eV for Pt})$ 
 $OH_{ad} + H_{ad} \rightarrow H_2O_{ad}$ 
 $(E_a = 0.11 \text{ eV for Pt}_3\text{Ni and } 0.11 \text{ eV for Pt})$ 

Figure 3 shows the potential energy surface of this preferred mechanism both on Pt and Pt<sub>3</sub>Ni surfaces. The rate-determining step (RDS) for this mechanism is the O hydration with a barrier of 0.34 eV for Pt<sub>3</sub>Ni and 0.29 eV for pure Pt. Hence, for the gas phase, Pt outperforms  $Pt_3Ni$ .

3.2.2. Gas Phase Barriers for  $Pt_3Ni$  at Higher Coverage. At higher coverage in the gas phase, the situation does not change significantly compared to the low coverage. The ORR mechanism remains the same except for the RDS which is now the water formation with a barrier of 0.36 eV, while at the low coverage it is the O hydration step with a slightly lower barrier of 0.34 eV (see Table 3).

3.2.3. Solvated Phase Barriers for  $Pt_3Ni$  at Low Coverage. O–O Bond Activation. The best  $O_2$  dissociation pathway starts from an  $f_2$ -bound  $O_2$  which dissociates to form two  $O_{ad}$  at  $f_2$  sites with no barrier (because solvent strongly favors dissociation), similar to that on pure Pt. We find that OOH formation has a barrier of 0.17 eV on  $Pt_3Ni$ , close to 0.19 eV on pure Pt. OOH dissociation is barrierless on  $Pt_3Ni$ , similar to that on Pt. Thus, the OOH formation and dissociation are not unfavorable pathways for the solvated system.

OH Formation. For the second step, Pt<sub>3</sub>Ni has a direct OH formation barrier of 1.02 eV, slightly lower than 1.09 eV on pure Pt. The O hydration step is less favorable on Pt<sub>3</sub>Ni, leading to a barrier of 0.81 eV, compared to 0.50 eV for pure Pt. Earlier,<sup>22</sup> we showed that the O hydration is the dominant mechanism for forming OH on Pt. Hence, the O hydration is the preferred mechanism for both Pt<sub>3</sub>Ni and Pt with a barrier of 0.81 and 0.50 eV, respectively. This suggests that Pt<sub>3</sub>Ni would have a worse performance for formation of OH<sub>ad</sub>.

OH Consumption. For the consumption of OH, water formation on  $Pt_3Ni$  with a barrier of 0.25 eV is slightly worse than that on Pt (the barrier is 0.17 eV).

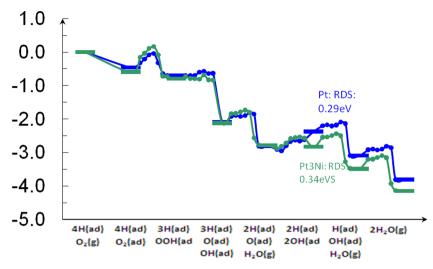


Figure 3. Potential energy surface including barriers for the OOH-form-hydr mechanism preferred for both Pt and Pt<sub>3</sub>Ni in the gas phase.

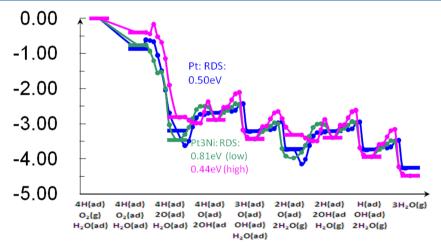


Figure 4. Potential energy surface including barriers for the  $O_2$ -diss-hydr mechanism preferred for both  $Pt_3Ni$  and Pt in solution. The purple line shows the alternative mechanism at higher coverage.

Summarizing the above discussion, we come to the conclusion that the  $O_2$ -diss-hydr-low mechanism is favorable in the solvated phase:

$$O_{2ad} \rightarrow 2O_{ad}$$
  $(E_a = 0.00 \text{ eV both for Pt}_3\text{Ni and Pt})$ 
 $O_{ad} + H_2O_{ad} \rightarrow 2OH_{ad}$ 
 $(E_a = 0.81 \text{ eV for Pt}_3\text{Ni and 0.50 eV for Pt})$ 
 $OH_{ad} + H_{ad} \rightarrow H_2O_{ad}$ 
 $(E_a = 0.25 \text{ eV for Pt}_3\text{Ni and 0.17 eV for Pt})$ 

Figure 4 shows the potential energy surface of the  $O_2$ -diss-hydr-low mechanism with solvent effects at different coverage. In solution, the O hydration is the RDS for both  $Pt_3Ni$  and Pt with the overall barrier of 0.81 and 0.50 eV, respectively. Thus, our computation suggests that  $Pt_3Ni$  would not outperform Pt. This result disagrees with the experimental observation, showing that  $Pt_3Ni$  is a more efficient ORR catalyst than pure Pt.

3.2.4. Solvated Phase Barriers for Pt<sub>3</sub>Ni at Higher Coverage. The above analysis is based on the assumption that all reactants are at the preferred sites, i.e., the reactants are within the orange region in Figure 1. In contrast, at higher coverage, the reactants

might end up in the blue region, even though it is not preferred at low coverage. The adsorbate coverage on the catalyst surface can range from 1/4 to as high as 2/3 monolayer.  $^{28-30}$  At higher coverage, the blue region becomes accessible for binding. Actually, it is not necessary for the coverage to be higher than 1/4 monolayer to trigger binding to the less preferred sites, because the large migration barrier may also block the rearrangement of O atoms toward adjacent sites (the O migration barrier between  $f_2$  sites is 0.6 eV). To consider the changes that might occur at higher coverage, we calculated the corresponding barriers for the blue region.

For the  $O_2$  dissociation starting from  $h_{1b}$ ,  $O_2$  can easily dissociate to form  $O_{ad}$  in the  $f_{1a}$  and  $f_{1b}$  sites with a barrier of 0.24 eV in solution and 0.92 eV in the gas phase.

For  $O_{ad}$  at the  $f_{1a}$  and  $f_{1b}$  sites, the corresponding O hydration reaction with nearby  $H_2O$  has a barrier of 0.44 eV in solution but is barrierless in the gas phase.

The following step of the  $\rm H_2O$  formation also has a low barrier of 0.43 eV in the solvated phase and 0.36 eV in the gas phase.

Hence, when the preferred region is covered with  $O_{ad}$ ,  $O_2$  can start to bind to the unfavorable  $h_{1b}$  site, and then dissociate to  $f_{1a}$  and  $f_{1b}$ . This allows a lower barrier for the hydration step,  $O_{ad} + H_2O_{ad} \rightarrow 2OH_{ad}$ , avoiding the high O hydration

and OH formation barriers calculated for low coverage, 0.81 and 1.02 eV, respectively. The proposed mechanism is  $O_2$ -diss-hydr-high:

$${
m O}_{2{
m ad}} 
ightarrow 2{
m O}_{{
m ad}}$$
 ( $E_{
m a}=0.00~{
m eV}$  at low coverage and 0.24 eV at high coverage)

$$O_{ad} + H_2O_{ad} \rightarrow 2OH_{ad}$$
  
 $(E_a = 0.81 \text{ eV} \text{ at low coverage and 0.44 eV at high coverage})$ 

$$OH_{ad} + H_{ad} \rightarrow H_2O_{ad}$$
  
(E<sub>2</sub> = 0.25 eV at low coverage and 0.43 eV at high coverage)

Figure 4 shows the potential energy surface of the  ${\rm O_2}$ -disshydr mechanism in solution at different coverages. The overall mechanism has the O hydration as the RDS with a barrier of 0.44 eV, slightly smaller than 0.50 eV for pure Pt. This mechanism occurs on the  ${\rm Pt_3Ni}(111)$  surface at higher coverage.

Thus, in the gas phase, the preferred mechanism is OOH-form-hydr with a RDS barrier of 0.34 eV for Pt<sub>3</sub>Ni and 0.29 eV for Pt. In solution at low coverage, where all intermediates can bind to the preferred region, we predict that Pt<sub>3</sub>Ni would have slower kinetics than Pt with an overall barrier of 0.81 eV. However, at higher coverage, the less preferred (blue) region might be involved in the process, which leads to a RDS barrier of 0.44 eV, smaller than that of 0.50 eV for pure Pt. This result is consistent with experiment.

3.3. Origin of Improved Performance. Comparing the RDS barriers for Pt and Pt<sub>3</sub>Ni, we can see that the major problem for both Pt and Pt<sub>3</sub>Ni alloys as ORR catalyst is the slow OH formation step. At low coverage, the situation becomes even worse, because of the stronger binding of O<sub>ad</sub> by 0.13 eV on Pt<sub>3</sub>Ni as compared to pure Pt. This makes any step that converts O<sub>ad</sub> into OH<sub>ad</sub> more difficult on Pt<sub>3</sub>Ni. However, at higher coverage or less organized coverage, when the favored sites are not all available, O atoms begin adsorbing onto the less favored blue region, which leads to a much lower reaction enthalpy and barrier for OH formation either via direct OH formation or O hydration. To reach better performance of Pt and Pt<sub>3</sub>Ni catalysts, the OH formation barrier should be lowered. This agrees with the volcano-like trend for ORR catalysts, <sup>7,31,32</sup> when the activity first increases and then decreases as the catalyst d-band center shifts downward. A lower d-band center provides stronger binding of OH<sub>ad</sub> and O<sub>ad</sub>. A strong enough binding of O atom is required for O adsorption, but too strong binding makes OH formation (and probably H2O formation) more difficult. Our study shows that Pt<sub>3</sub>Ni has, on average, weaker O and OH binding, especially on the blue region. This is in agreement with earlier obtained results.<sup>7,33</sup>

From our calculation, we find more difficult water formation on  $\text{Pt}_3\text{Ni}$  compared to pure Pt, which is a much smaller issue when  $\text{H}^+$  is involved in the reaction.  $^{13,27,34}$  A weaker bonded OH would be easier removed by  $\text{H}^+$ . This is consistent with the assumption about a spatial effect;  $^{5,35}$  easier OH removal leads to more surface area for O adsorption. However, our results show that the more difficult step for  $\text{Pt}_3\text{Ni}$  is  $\text{O} \to \text{OH}$ ,  $E_a = 0.80$  and 0.44 eV via OH formation and O hydration, respectively, at high coverage (see Table 4) rather than the OH  $\to$  H<sub>2</sub>O step,  $E_a = 0.25-0.43$  eV via direct formation and becomes negligible if  $\text{H}^+$  is involved in the reaction.  $^{34}$  These barriers are lower than the corresponding barriers for

Pt ( $O_{ad} \rightarrow OH_{ad}$ ,  $E_a$  = 1.09 eV via direct OH formation and  $E_a$  = 0.50 eV via O hydration). The lower barriers lead to the improved performance.

#### 4. CONCLUSIONS

We studied systematically the binding site preference of all reaction intermediates involved in ORR on  $Pt_3Ni$ . The binding energy of adsorbates on the alloy surface shows the strong sublayer dependence. Compared with the pure Pt surface, the binding sites are partitioned into two regions, the preferred (orange) and the less preferred (blue) region. The mechanism of ORR on  $Pt_3Ni$  is similar to that on pure Pt. In solvent,  $O_2$  first dissociates into  $O_{adv}$  then  $O_{ad}$  reacts with  $H_2O$  to produce  $OH_{adv}$  which finally reacts with  $H_{ad}$  to form  $H_2O$ .

Due to the substantial difference in binding sites, ORR shows different kinetics for the preferred and less preferred regions. The overall barrier for ORR at the preferred region is 0.81 eV, while the barrier for the less preferred region is significantly lower, 0.43 eV. Thus, the ORR activity should be strongly coverage-dependent. At low coverage, all reactants and intermediates adsorb only onto the preferred region, generating a slower reaction rate than that on pure Pt. At higher coverage,  $\rm O_2$  can react at the less preferred region, leading to a better rate as compared to pure Pt.

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#### Notes

The authors declare no competing financial interest.

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