

# Carboxylic Acid-Directed Clustering and Dispersion of $\text{ZrO}_2$ Nanoparticles in Organic Solvents: A Study by Small-Angle X-ray/Neutron Scattering and NMR

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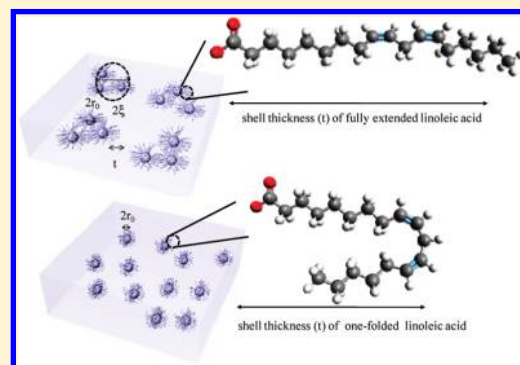
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**S** Supporting Information

**ABSTRACT:** The stable dispersion of nanoparticles in organic solvents, essential for their practical application, is often achieved by the grafting of organic dispersants. Good dispersion is usually assumed when a transparent colloidal sol is formed. However, even in such a case, the dispersion may not be in the form of distinct nanoparticles. In this study, zirconia nanoparticles have been modified with a series of carboxylic acids so that they can be dispersed in polar protic/aprotic solvents or nonpolar organic solvents as transparent colloidal sols. The dispersed state in these solvents was characterized by SAXS and SANS. In most cases, the transparent sol consists of “soft” secondary clusters generated from the  $\text{ZrO}_2$  nanoparticles. The size and fractal structure of the dispersed clusters varied with the acid modifier. Only in the linoleic acid-modified case and with chloroform as the solvent was the  $\text{ZrO}_2$  dispersed as primary particles. However, the same modification leads to secondary clusters in other solvents, such as toluene and benzene, with similar solubility parameters. The difference in the dispersion states calls for a molecular-level interpretation. It was proposed that the grafted LOA exists as a swollen and extended brush in benzene and toluene, but as a folded and compact shell in chloroform. The proposed shell structures were then confirmed by the proximity of the chain tail with the middle section observed on 2D NOESY  $^1\text{H}$  NMR spectra.



## INTRODUCTION

The incorporation of inorganic nanoparticles is one of the effective ways to modulate the refractive index of an optical polymer. Among the possible nanofillers,  $\text{ZrO}_2$  is preferred for its high transparency in both the UV and the visible ranges. Nanocomposites of  $\text{ZrO}_2$  nanoparticles with a variety of acrylate photopolymers have been studied as materials for holographic recording,<sup>1–3</sup> LED encapsulation,<sup>4</sup> and other optical functional materials<sup>5–7</sup> or coatings.<sup>8–10</sup>

$\text{ZrO}_2$  nanoparticles can be prepared by many different processes. For the preparation of optical nanocomposites, the particle size must be much smaller than 10 nm so that excessive Rayleigh scattering is avoided. Furthermore, the hydrophilic surface of  $\text{ZrO}_2$  must be modified to be compatible with organics, thus intimately mixed with the polymer. Many different capping agents have been employed for such a purpose. Among them, the silane coupling agents are most popular, particularly the one with a methacrylic group. Carboxylic acids are also favored for not introducing unnecessary siloxane bridges.

After surface modification, the nanoparticles are usually dispersed in a solvent before mixing with the desired monomers or

oligomers. For optical applications where transparency is critical, a uniform distribution of the nanoparticles in the organic phase is essential. Thus, it is important to keep the nanoparticles from aggregating into a larger size. However, it is known that nanoparticles, even after surface modification, seldom disperse in solvent as primary units. The competition and balance between short-range attraction and long-range repulsion may yield a variety of anisotropic superstructures (clusters or aggregates).<sup>11,12</sup> The mixture of nanoparticles and oligomers in solvent might still be transparent as long as the size and refractive index of the composite clusters or aggregates are below the scattering limit, but the rheology<sup>13–15</sup> of the mixed compositions and the mechanical properties<sup>16</sup> of the final nanocomposite will be greatly influenced by the shape, size, structure, and distribution of the clusters.

Dynamic light scattering (DLS) and transmission electron microscopy (TEM) had been the main tools to study the dispersion

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of surface-modified nanoparticles in solvents.<sup>17–20</sup> However, DLS gives only the cluster size, but not its structure. TEM analysis is made on dried samples, and is strongly affected by the sample preparation procedures. Comparing to the above tools, small-angle X-ray and neutron scattering can be done in situ and give more information concerning the nanometer-scale structure of the dispersed phase.

In this work, the dispersion behavior of carboxylic acid-modified zirconia nanoparticles in various protic/aprotic and nonpolar organic solvents is studied by both small-angle X-ray scattering (SAXS) and small-angle neutron scattering (SANS). The nanoparticles were dispersed as primary units in only a single combination of modifier (linoleic acid) and solvent (chloroform). In all other cases, the modified particle formed “soft” secondary clusters whose size and fractal structure varied with the acid modifier. The peculiar behavior of linoleic acid was attributed to the conformation of the organic shell in chloroform. This proposal was further substantiated using <sup>1</sup>H NMR.

## EXPERIMENTAL SECTION

**Materials.** Redispersible ZrO<sub>2</sub> nanocrystals were produced from the low-temperature (110 °C) hydrothermal reaction of zirconium hydroxide nanoparticles produced by hydrolysis at a fixed pH value.<sup>21</sup> Briefly, 64.4 g of ZrOCl<sub>2</sub>·8H<sub>2</sub>O was dissolved in 58 g of DI water, to which 2.1 g of methacrylic acid was added. The mixture was stirred for 30 min at RT to dissolve the ZrOCl<sub>2</sub> until a pale green transparent solution resulted. The MA added to the ZrOCl<sub>2</sub>·8H<sub>2</sub>O solution was to protect the zirconium polycation and to block the formation of large clusters.

The zirconium precursor was then heated at 110 °C for 24 h before being hydrolyzed in a buffer solution with 4 M NaOH. The buffer solution consists of 10 g of NH<sub>4</sub>HCO<sub>3</sub> and 5.5 g of triethanolamine (TEA) in 800 g of DI water. The hydrolysis was done at a fixed pH value of 10 by controlling the addition rates of the acid and base. The buffered mixture remained clear during the whole neutralization process if the neutralization step lasted for more than 4 h. The hydrolysis product was collected by flocculating with acid (HCl) at a pH ~ 5.4. The excess electrolytes were washed away by repeated centrifugation. IR analysis of the washed precipitates obtained indicated the absence of MA, but the existence of TEA. This was because the hydrolysis was done at pH = 10, which is well above the iso-electric point (IEP) of zirconium hydroxide. The surface of the hydrated hydroxide nanoparticles produced would be negatively charged and repel the MA anion, but attract the TEA cation.

In the final step, the washed hydroxide product was dispersed in water and adjusted to 6 wt % and a pH value of 11.5, then subjected to hydrothermal reaction at 110 °C for 15 h. The reaction product was centrifuged, with the interstitial water replaced by acetone, then vacuum-dried into powders. The powders were redispersible in water, leading to a transparent sol with pH ~ 10.

A series of carboxylic acids (acetic, acrylic, methacrylic, propionic, butyric, and linoleic) were employed as the surface modifier. The precipitates were collected by centrifugation and vacuum-dried into fluffy powders. The products are denoted as Ac–, AA–, MA–, PA–, BA–, and LOA–ZrO<sub>2</sub> in the same sequence. When modifying the ZrO<sub>2</sub> with carboxylic acid, the pH value must always be kept below the IEP so that chelating is possible (see the Supporting Information, Figure S1). For example, MA modification was conducted by adding these

**Table 1.** Goodness of Dispersion for Carboxylic Acid-Modified ZrO<sub>2</sub> Nanoparticles in Various Solvents<sup>a</sup>

solvent	acid					
	Ac	PA	AA	BA	MA	LOA
H <sub>2</sub> O	o	x	x	x	x	x
EtOH	o	x	x	x	o <sup>b</sup>	x
THF	x	o	o	o	o	x
chloroform	x	o	o	o	o	o
hexane	x	x	x	x	x	o
benzene	x	x	x	o	o	o
toluene	x	x	x	o	o	o
MMA	x	x	x	o	o	x

<sup>a</sup> By visual inspection at a 10 wt % loading of ZrO<sub>2</sub>: o, dispersible as transparent sol; x, precipitation or flocculation. <sup>b</sup> Incompletely dried.

powders into an aqueous MA solution at a pH value of about 4.3–4.6 and stirring at 60 °C for 2 h. The precipitates were centrifuged and repeatedly washed by water and vacuum-dried at 80 °C to obtain modified powders. LOA modification was conducted similarly by adding the powders into an LOA/ethanol solution at pH = 6 and stirring at 60 °C for 2 h. The precipitates were centrifuged and repeatedly washed with ethanol and vacuum-dried at 30 °C to obtain the LOA-modified powders. TGA analysis (Figure S2, Supporting Information) showed that the organic loading was 28, 32, 32, 35, 35, and 50 wt % for Ac–, AA–, PA–, MA–, BA–, and LOA–ZrO<sub>2</sub>, respectively.

The modified powders could be dispersed in various organic solvents as transparent sols. The nanoparticle dispersions were always at 10 wt % in solid, unless otherwise mentioned. The judgment on a good dispersion was made when the dispersion appeared as transparent or slightly bluish translucent. In such a case, the sol remains transparent (translucent) for at least 3 months. The transparency of the sol can of course be defined more clearly by a transmission spectrum, as given by the example in Figure S3 (Supporting Information). The goodness of dispersion, as observed visually, is tabulated in Table 1.

**SAXS and SANS.** Various sols, with 0.1–1 wt % ZrO<sub>2</sub>, were subjected to SAXS measurements. The SAXS experiments were performed at beamline BL23A of the National Synchrotron Radiation Research Center (NSRRC), Hsinchu, Taiwan. The configuration of the SAXS at BL23A was detailed elsewhere.<sup>22</sup> A monochromatized X-ray radiation source with an energy of 12 keV (wavelength,  $\lambda = 1.03$  Å) was employed, and 2D SAXS patterns were typically collected for 180–300 s with a Mar165 CCD detector (Mar USA Inc., Evanston, IL, USA) with a 512 × 512 pixel array. The sample-to-detector distance was 2520.17 mm. After data reduction with homemade software, profiles of the scattering intensity ( $I$ ) versus scattering vector,  $Q$  ( $Q = 4\pi \cdot \sin(\theta/2)/\lambda$ , where  $\theta$  is the scattering angle), were obtained. The calibration of the detector pixels in terms of scattering vector was made with a silver behenate standard, whose  $Q_{\text{max}}$  was  $0.1076 \text{ Å}^{-1}$ . To remove the background scattering from solvents, air, and sealed kapton windows, precise measurements on the transmission of ZrO<sub>2</sub> dispersed sols and the corresponding organic solvents were also made. A 1.89 mm thick polyethylene (PE) block was used as the calibration standard, to convert the 1D scattering profiles to absolute intensities after background subtraction. The intensity of the diffraction peak of PE,  $I(Q) = 36.6 \text{ cm}^{-1}$

at  $Q = 0.023 \text{ \AA}^{-1}$ , was used as the scaling factor between the measured  $I(Q)$  and the absolute  $I(Q)$ .

A more concentrated dispersion (10 wt % in solvents) was employed in the SANS studies. SANS experiments were carried out at the QUOKKA end-station of the Australian Nuclear Science and Technology Organisation (ANSTO) in Australia. Samples were filtered (0.45  $\mu\text{m}$  membrane) and placed in a homemade demountable 1 mm sample cell sealed by quartz windows. A neutron wavelength of 4.9  $\text{\AA}$  with a 6% wavelength spread (full width at half-maximum) was used. Two sample-to-detector distances (13 and 2 m) were selected to provide a  $Q$  range of 0.06–0.3  $\text{\AA}^{-1}$ . Longer distances were run for 30 min to record the low  $Q$  values, whereas the shorter distances were run for 15 min. To perform background subtraction, the corresponding blank measurements with pure solvents were also made. The 2D SANS data, after subtracting the background, were normalized to a standard plexiglas sample to correct for detector efficiency. The corrected and normalized 2D SANS patterns were subsequently integrated to obtain 1D profiles of absolute intensity. All data reduction steps were performed using the NIST macros in Igor Pro using Quokka specific modifications.<sup>23</sup> The experimental scattering profiles were fitted by the Igor Pro using simulated scattering models developed by S. R. Kline.<sup>24</sup>

**NMR.** Proton NMR spectra (500.13 MHz) were collected with a Bruker Avance DMX500 spectrometer. The samples include 10 wt % LOA dissolved in deuterated chloroform and benzene, as well as LOA–ZrO<sub>2</sub> dispersions in the same set of solvents. Two-dimensional nuclear Overhauser enhancement spectroscopy (2D NOESY) spectra were recorded with a 700 ms mixing time and a recovery delay of 2 s. 2K data points were collected for 512 increments of 8 scans, using TPPI  $f_1$  quadrature detection.

**Data Analysis and Model Fitting.** SAXS data were analyzed by Guinier fitting, power-law fitting, and Beaucage model fitting.<sup>25</sup> For a colloidal dispersion, the scattering intensity can be approximated in the  $Q < 1/R_g$  range as<sup>26</sup>

$$\ln I(Q) = \ln I(0) - (Q^2 R_g^2)/3 \quad (1)$$

where  $R_g$  is the radius of gyration of primary nanoparticles or clusters. In the high- $Q$  region, the intensity decays as a function of  $Q$  with a power of  $-D_f$ , given by<sup>26</sup>

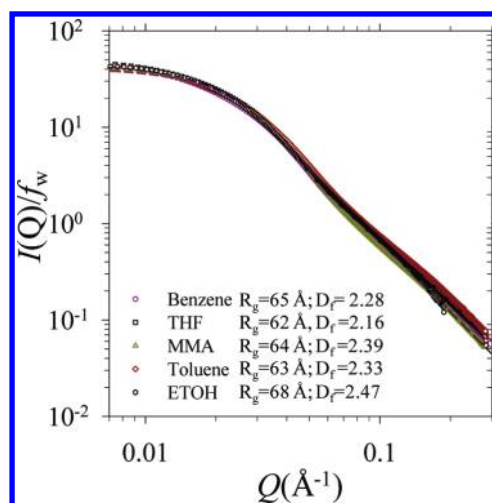
$$I(Q) = A Q^{-D_f} + B \quad (2)$$

where  $A$  is a constant,  $B$  is the incoherent background, and  $D_f$  the power-law exponent. For SAXS data, the incoherent background is subtle and generally set as zero when doing power-law fitting.

When the cluster size is large and beyond the Guinier domain that SAXS can measure, the Guinier approximation given by eq 1 may be invalid. To calculate the size of clusters in a solvent, the Beaucage model given by<sup>25</sup>

$$I(Q) = G \exp\left(-\frac{Q^2 R_g^2}{3}\right) + B \left\{ \frac{\left[ \text{erf}\left(\frac{Q R_g}{\sqrt{6}}\right) \right]^3}{Q} \right\}^P \quad (3)$$

was used, where the first term on the right-hand side represents the Guinier form at low  $Q$  and the second term denotes the power-law form at high  $Q$ . In here,  $G = n^2 N_p I_e$  is the exponential



**Figure 1.** Weight-fraction ( $f_w$ )-normalized experimental SAXS profiles (symbols) of MA–ZrO<sub>2</sub> dispersed in various organic solvents. The dashed lines represent the fitted profiles based on the Beaucage model with the parameters listed in the figure.

prefactor, where  $n$  is the number of electrons within a particle,  $N_p$  is the number of particles within a scattered volume, and  $I_e$  is the scattering factor of an electron. For Porod's law,  $B = 2\pi N_p \rho_s^2 S_p I_e$ , where  $\rho_s$  is the number density and  $S_p$  is the surface of the particle.<sup>25</sup>

For the SANS studies, the scattering from the grafted organic materials is not negligible. Thus, the SANS data were analyzed using a core–shell model composed of a polydisperse core with a constant shell thickness. For spheres with a core–shell structure<sup>27</sup>

$$P(Q) = \frac{\text{scale}}{V_t} \left[ \frac{3V_c(\rho_c - \rho_s)j_1(Qr_c)}{Qr_c} + \frac{3V_t(\rho_s - \rho_{\text{solvent}})j_1(Qr_t)}{Qr_t} \right] + bkg \quad (4)$$

where  $j_1 = (\sin r - r \cos r)/r^2$  is the first-order Bessel function,  $r_t = r_c + t_s$  and  $V_i = (4\pi/3)r_i^3$  represent the radius and volume, and the subscripts  $c$  and  $s$  represent the core and shell, respectively.  $t_s$  is the thickness of the shell. The Schultz distribution can be incorporated into eq 4 to give a polydispersity in the core radius.<sup>28</sup>

When surface-modified nanoparticles with a polydisperse core–shell structure aggregate into a fractal structure with a fractal dimension  $D_f$  and a correlation length  $\xi$ , the fractal model can be incorporated into eq 4 to fit the SANS intensity. We have

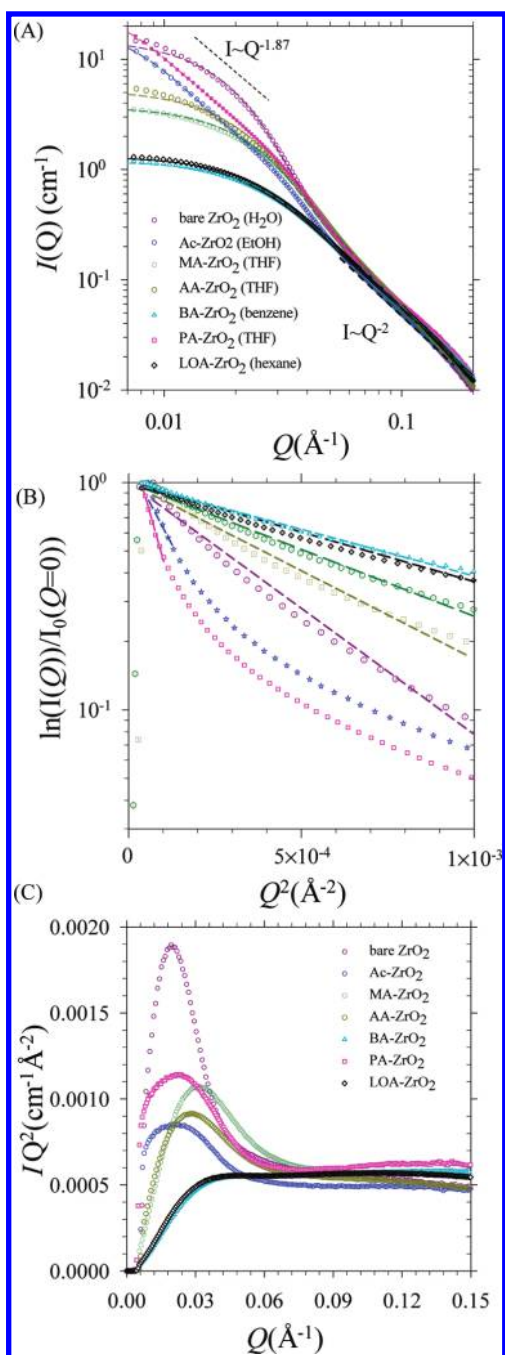
$$S(Q) = 1 + \frac{\sin[(D_f - 1)\tan^{-1}(Q\xi)]}{(Qr_0)^{D_f}} \frac{D_f \Gamma(D_f - 1)}{\left[ 1 + \frac{1}{Q^2 \xi^2} \right]^{(D_f - 1)/2}} \quad (5)$$

where radius  $r_0$  is for the primary nanoparticle that builds the fractal structure.<sup>29</sup>

## RESULTS AND DISCUSSION

**SAXS Characterization.** According to the analysis in our previous paper,<sup>21</sup> the crystalline ZrO<sub>2</sub> nanoparticles produced





**Figure 2.** (A) SAXS intensity versus wave vector data for surface-modified ZrO<sub>2</sub> (0.1 wt %) dispersed in different solvents. (B) Guinier plot of the same SAXS data set. (C) Kratky plot ( $IQ^2$  vs  $Q$ ) of the same SAXS data set.

had a BET surface of 177 m<sup>2</sup>/g and a grain size of ~11 nm. The powders were dispersible in water if modified with citric acid and tartaric acid, and in chloroform and hexane when modified with oleic acid. Since then, the preparation procedures have been further improved. Particularly, the washing after hydrothermal crystallization was more thorough and a vacuum-drying step was added after the carboxylic acid modification. Before surface modification by carboxylic acids, the bare ZrO<sub>2</sub> is dispersible (almost transparent at 10 wt %) in water. That modified with various carboxylic acids can be dispersed in polar protic (alcohols),

**Table 2.** Radii of Gyration and the Fractal Dimensions of the Clusters Obtained from Fitting the Data Shown in Figure 2A to the Beaucage Model<sup>a</sup>

modifier	unmodified	Ac—	PA—	AA—	MA—	BA—	LOA—
solvent	H <sub>2</sub> O	EtOH	THF	THF	THF	benzene	hexane
$R_g$ (Å)	94	55	44	79	62	54	57
$D_f$	2.47	2.15	2.17	2.45	2.16	2.34	2.47
$R_g^*$ (Å)		213	232				
$D_f^*$		1.88	1.87				

<sup>a</sup>The superscript \* represents the size and fractal dimension of the diffusion-limited aggregates grown from the Ac— or PA—ZrO<sub>2</sub> clusters.

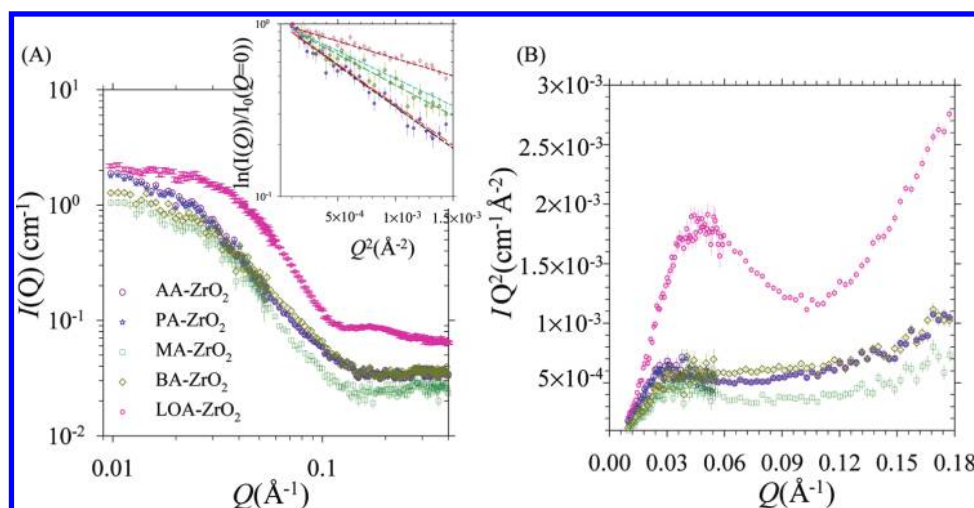
aprotic, and nonpolar organic solvents, as shown in Table 1. The MA—ZrO<sub>2</sub> and BA—ZrO<sub>2</sub> are particularly interesting, because they can be dispersed in most solvents tested, ranging from protic/aprotic to nonpolar ones, as well as methyl methacrylate (MMA) monomer. In contrast, that modified with smaller acids are preferably dispersed in polar protic/aprotic solvents, whereas that with long-chain acids favors the nonpolar organic solvents.

Diluted dispersions of MA—ZrO<sub>2</sub> (at 0.1 or 1 wt %) in various solvents were first studied by SAXS. The experimental scattering profiles and the fitted curves based on the Beaucage model are shown together in Figure 1. The experimental data indicate that the structure of the MA—ZrO<sub>2</sub> is independent of the solvent they are dispersed in. The radii of gyration estimated from the curve-fitting were in the range of 62–68 Å, and the fractal dimension  $D_f$  ranged from 2.16 to 2.47. Both suggest that the MA—ZrO<sub>2</sub> was dispersed as clusters in these organic solvents.

The structural details of the carboxylic acid-modified ZrO<sub>2</sub> nanoparticles (at 0.1 wt %) were also studied with SAXS. The results are presented in Figure 2A, where data on the aqueous dispersion of unmodified ZrO<sub>2</sub> nanoparticles are also included. The behaviors of the scattering intensity profiles can be divided into two regions. The absolute intensity at low  $Q$  varies with the modifier, but merges to a single curve at high  $Q$ . The intensity at low  $Q$  seems to decrease as the length of the carboxylic acid increases. For the BA—ZrO<sub>2</sub> and the LOA—ZrO<sub>2</sub> cases, the intensity reaches a plateau at the low- $Q$  limit. For the Ac—ZrO<sub>2</sub> and the PA—ZrO<sub>2</sub> cases, the scattering intensity decays with approximately  $I \sim Q^{-1.78}$ . This power-law decay at low  $Q$  indicates the development of very large and loose aggregates through the diffusion-limited cluster–cluster aggregation.<sup>30</sup> Because the size of these aggregates is beyond the Guinier domain SAXS can measure, it cannot be estimated by the Guinier approximation. On the other hand, all data except that with acetic acid and propionic acid are linear on the Guinier plots shown in Figure 2B, and their cluster sizes can be estimated by the slope.

For the Ac—ZrO<sub>2</sub> and the PA—ZrO<sub>2</sub> cases, the size of the large aggregates must be estimated by fitting the scattering profile to the Beaucage model (eq 3). The results obtained from fitting were also tabulated in Table 2. For consistency, all other data were also fitted to the same model to extract the corresponding fractal dimensions and the cluster sizes.

According to Table 2, both the Ac—ZrO<sub>2</sub> and the PA—ZrO<sub>2</sub> clusters grew into rather large aggregates, with  $R_g^* \sim 213$  Å and  $D_f = 1.88$  for the former and  $R_g \sim 236$  Å and  $D_f = 1.87$  for the latter. The low fractal dimension in both cases indicates that they are rather loosely linked. All other cases in this table were dispersed in the form of individual clusters. The unmodified ZrO<sub>2</sub> has a fractal dimension of  $D_f = 2.47$  and a cluster size of



**Figure 3.** (A) Neutron scattering intensity for carboxylic acid-modified  $\text{ZrO}_2$  dispersed in chloroform. The inset shows the Guinier plot. (B) Kratky plot ( $IQ^2$  vs  $Q$ ) for the same SANS data set.

95 Å. The cluster size reduced to  $\sim 60$  Å when modified with acids, but the fractal dimension remained at a similar value. The details of these cluster structures are more easily differentiated from the corresponding Kratky plot ( $IQ^2$  versus  $Q$ ) shown in Figure 2C. Here, the unmodified  $\text{ZrO}_2$  nanoparticles exhibit a sharp peak at low  $Q$ , whereas BA- or LOA- $\text{ZrO}_2$  only leads to a plateau. The former is a typical characteristic of spherical aggregates, whereas the latter resembles the Gaussian soft spheres.<sup>13,31,32</sup> The clusters formed by MA- $\text{ZrO}_2$  should be somewhere in between.

**SANS Characterization.** As shown in Table 1, all modified nanoparticles except Ac- $\text{ZrO}_2$  can be dispersed in chloroform. Because the X-ray intensity is significantly attenuated by the heavy chlorine element, the chloroform sols are unsuitable for SAXS measurement. Consequently, SANS was used to characterize the structural details of the dispersion in chloroform. Shown in Figure 3A are the obtained data.

In this figure, the intensity falls exponentially according to the Guinier approximation at the low- $Q$  region ( $0.01\text{--}0.03\text{ Å}^{-1}$ ). Within this  $Q$  range, the scattering feature is provided by the primary nanoparticles or the secondary clusters whose size is within the Guinier range. The Guinier curves gathered in the inset reveal that the radius of gyration for the clusters seems to decrease with increasing length of the acid. However, unlike the SAXS data where the intensity decayed with a power-law at very high  $Q$ , the SANS curves reveal a plateau-like feature at the same  $Q$  range ( $0.1\text{--}0.4\text{ Å}^{-1}$ ). The high- $Q$  plateau is the result of both coherent and incoherent scattering of the grafted carboxylic acids. Particularly, the SANS profile of LOA- $\text{ZrO}_2$  reveals a broad hump around  $Q = 0.177\text{ Å}^{-1}$ , a signature of a core-shell structure.

Figure 3B shows the corresponding Kratky plot ( $IQ^2$  versus  $Q$ ) of the same data. All curves exhibit a high- $Q$  plateau, except that for the LOA- $\text{ZrO}_2$  case. Such a feature indicates the formation of Gaussian soft spheres.<sup>31,32</sup> This phenomena was also seen by Mackay et al.<sup>13</sup> for tightly cross-linked PS nanoparticles. In our colloidal system, the clusters comprise an inorganic core and a grafted carboxylic acid. The density profile within the clusters is not uniform, thus leading to a plateau in the Kratky plot.

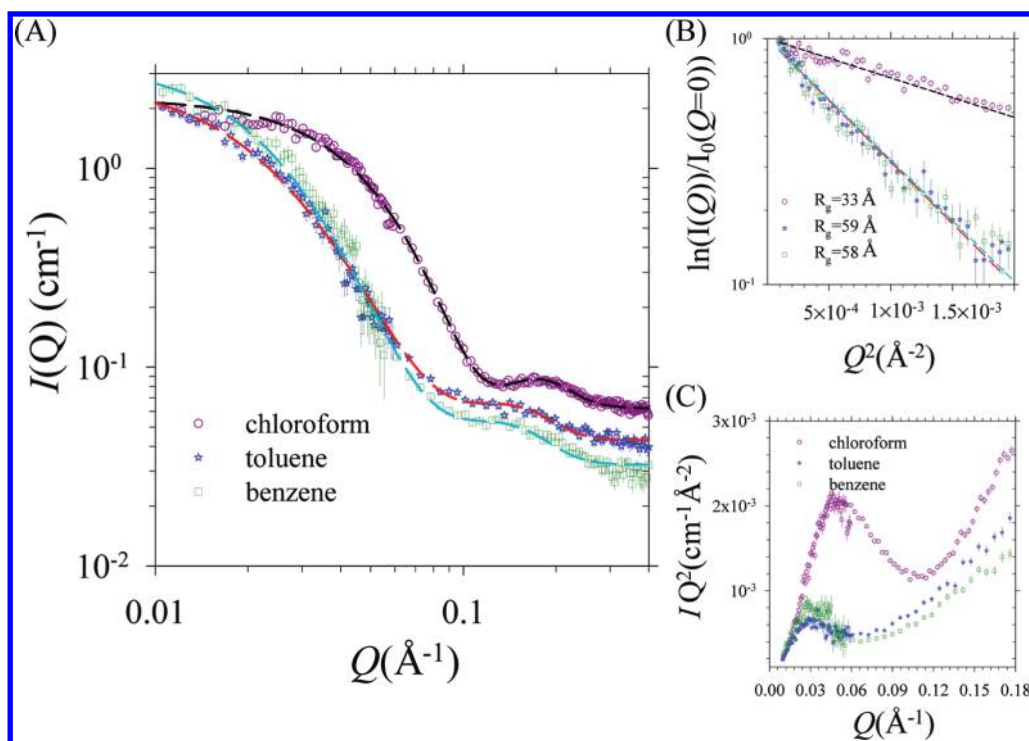
The Kratky plot for the LOA- $\text{ZrO}_2$  system, on the other hand, clearly reveals a single broad peak centered at  $Q = 0.04\text{ Å}^{-1}$  without the presence of a high- $Q$  plateau. Such a peak was not

seen in the SAXS data of the same sample when dispersed in hexane (Figure 2C). The Kratky plot for LOA- $\text{ZrO}_2$ /chloroform is very similar to that observed for a hard sphere with a homogeneous density.<sup>31</sup> The most likely explanation is that LOA- $\text{ZrO}_2$  primary particles are homogeneously dispersed in chloroform without clustering.

Combining the SAXS and SANS data, we found that there are two types of dispersion behaviors for the same LOA- $\text{ZrO}_2$  sample. They exist as primary particles when dispersed in chloroform and as secondary clusters when dispersed in other solvents. This requires some reasoning. One possibility is that the conformation of the grafted linoleic acid may change in different solvents, thus affecting the dispersion state. Linoleic acid is an unsaturated fatty acid, comprising an 18-carbon chain with two double bonds. Such a long chain is expected to have diverse conformations in the solvent.

To probe the conformation of grafted linoleic acid in different solvents, additional SANS measurements were made on 10 wt % dispersions of LOA- $\text{ZrO}_2$  in toluene and benzene. These data were compared with those collected with chloroform as the solvent in Figure 4. All SANS curves in Figure 4A exhibit a common contribution from the scattering of the  $\text{ZrO}_2$  core (at low- $Q$  region) as well as that of the grafted linoleic acid (contributing to the broad hump at  $Q > 0.1\text{ Å}^{-1}$ ). Again, the power-law intensity decay ( $I \sim Q^{-D_f}$ ) gives the fractal exponent in the range of  $Q = 0.03\text{--}0.08\text{ Å}^{-1}$ . The exponent  $D_f$  was 2.86 with chloroform as the solvent, 2.74 with benzene, and 2.33 with toluene. This is to say that the LOA- $\text{ZrO}_2$  had a more compact structure in chloroform than in toluene or benzene. The exponential decay of the SANS intensity at a small  $Q$  range follows the Guinier approximation, as demonstrated in Figure 4B. The radius of gyration, inversely proportional to the slope, is much smaller in the chloroform case compared with that with toluene or benzene as the solvent. Note that the hydrodynamic diameter of LOA-grafted  $\text{ZrO}_2$  dispersed in chloroform is also the smallest, identified by dynamic light scattering (DLS) (see the Supporting Information, Figure S4).

We now examine the high- $Q$  region ( $Q > 0.1\text{ Å}^{-1}$ ) in Figure 4A. Here, the scattering intensity reveals a broad hump that was absent in the SAXS profiles collected with toluene and benzene as solvents (see the Supporting Information, Figure S5).



**Figure 4.** (A) Neutron scattering intensity data for LOA–ZrO<sub>2</sub> dispersed in chloroform, toluene, and benzene. Dashed curves were obtained by model fitting. Guinier plot (B) and Kratky plot (C) for the same SANS data set.

**Table 3. Results of Fitting the LOA–ZrO<sub>2</sub> SANS Data with the Fractal Polydisperse Core–Shell Structure Model**

parameters	chloroform	toluene	benzene
block radius (Å)	12.4	10.5	12.1
fractal dimension	2.86	2.33	2.74
shell thickness (Å)	14.1	18.8	17.5
polydispersity	0.263	0.33	0.32
correlation length (Å)	14.1	42.7	35.1

The linoleic acid, containing only light elements, has an X-ray scattering length density of  $8.52 \times 10^{-6} \text{ \AA}^{-2}$ , similar to that of the solvent (toluene,  $8.0 \times 10^{-6} \text{ \AA}^{-2}$ ; benzene,  $8.02 \times 10^{-6} \text{ \AA}^{-2}$ ) and much smaller than that of ZrO<sub>2</sub> ( $4.31 \times 10^{-5} \text{ \AA}^{-2}$ ). Therefore, the X-ray is practically “blind” to the grafted linoleic acid against the solvent background. Comparatively, the neutron is much more sensitive. Thus, the extrabroad hump in the SANS profile reflects the local structure details of the organic shell.

Notice that a larger slope on the Guinier plot, or the radius of gyration, was observed when the same sample is dispersed in toluene or benzene than that in chloroform. This clearly indicates the clustering of particles in the first two solvents. To fit the SANS data, a fractal polydisperse core–shell model was implemented. During the fitting, the fractal dimension of the clusters was fixed to the values previously identified by a power-law analysis, that is,  $D_f = 2.86$ ,  $2.74$ , and  $2.33$  for cases with chloroform, benzene, and toluene, respectively. The neutron scattering length density of ZrO<sub>2</sub> was taken as  $\rho_{\text{ZrO}_2}^n = 5.21 \times 10^{-6} \text{ \AA}^{-2}$ . That of solvents and linoleic acid were taken as  $\rho_{\text{chloroform}}^n = 2.39 \times 10^{-6} \text{ \AA}^{-2}$ ,  $\rho_{\text{toluene}}^n = 9.4 \times 10^{-7} \text{ \AA}^{-2}$ ,  $\rho_{\text{benzene}}^n = 1.18 \times 10^{-6} \text{ \AA}^{-2}$ , and  $\rho_{\text{LOA}}^n = 2.24 \times 10^{-7} \text{ \AA}^{-2}$ . The results have been plotted as dashed lines in Figure 4A to demonstrate the goodness

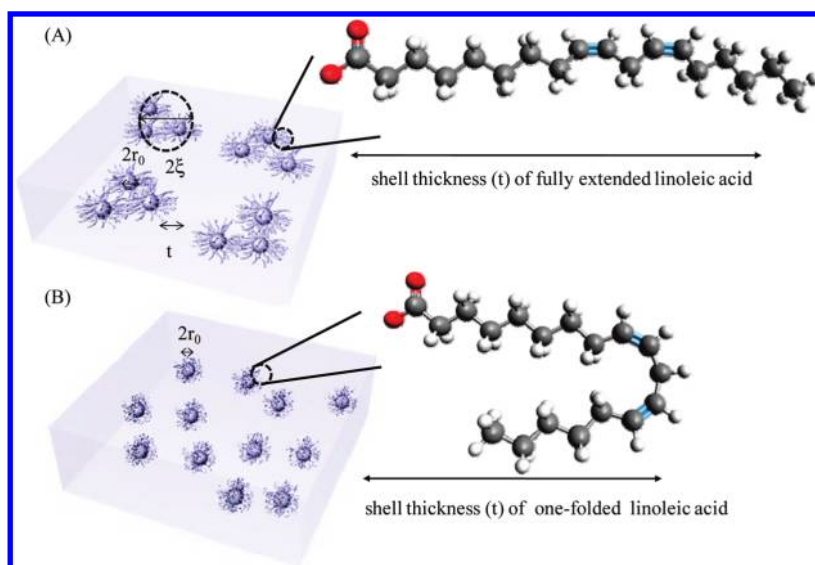
of fit, whereas the structural parameters obtained are summarized in Table 3.

The clusters in toluene and benzene exhibit a blocking (core) radius of  $10.5$ – $12 \text{ \AA}$ , with an organic shell of  $17.5$ – $18.8 \text{ \AA}$  in thickness and a correlation length of  $35$ – $43 \text{ \AA}$ . Although LOA-grafted ZrO<sub>2</sub> exists as loose, small clusters in C<sub>6</sub>H<sub>6</sub>, interpenetration of LOA chains among different LOA-grafted ZrO<sub>2</sub> in a cluster can be excluded. Furthermore, the cluster is 3 or 4 times the size of the primary particle. Therefore, each cluster only comprises several primary particles. Within a cluster, very little surface area of each primary particle is in contact with that of its neighbors. On average, for primary particles within the cluster, the shell composed of grafted LOA still exhibits a single monolayer thickness. The same LOA–ZrO<sub>2</sub> in chloroform has a blocking (core) radius similar to the other cases, but with a thinner shell and a much shorter correlation length. The correlation length is only  $14.1 \text{ \AA}$ , comparable to the core size. This is a strong indication that the chloroform dispersion is made of primary particles with a compact core.

The fitting results can be best represented by the Kratky plot (Figure 4C) where a characteristic maximum for a particle-like dispersion<sup>13,31,32</sup> is observed. The larger and softer clusters formed in toluene or benzene produced a much weaker maximum and at a smaller  $Q$  value compared to that in chloroform. For the latter case, the upward trend at high  $Q$  is also sharper. This indicates the strong neutron scattering from more or less rigid entities at a very small scale. This must have come from the thinner and more compact conformation of the organic shell in chloroform than that in toluene or benzene.

The conformation of a solute is often related to the solubility parameter of the solvent. However, the solubility parameters for these three solvents [ $\delta_{\text{chloroform}} = 9.3$ ,  $\delta_{\text{toluene}} = 8.9$ , and  $\delta_{\text{benzene}} = 9.2$  ( $\text{cal}/\text{cm}^3$ )<sup>1/2</sup>]<sup>33</sup> are very similar, which cannot explain our





**Figure 5.** Schematic illustration of (A) the LOA-ZrO<sub>2</sub> clusters dispersed in toluene and benzene. The grafted LOA chains are fully extended in both solvents. (B) The LOA-ZrO<sub>2</sub> primary particle dispersed in chloroform.

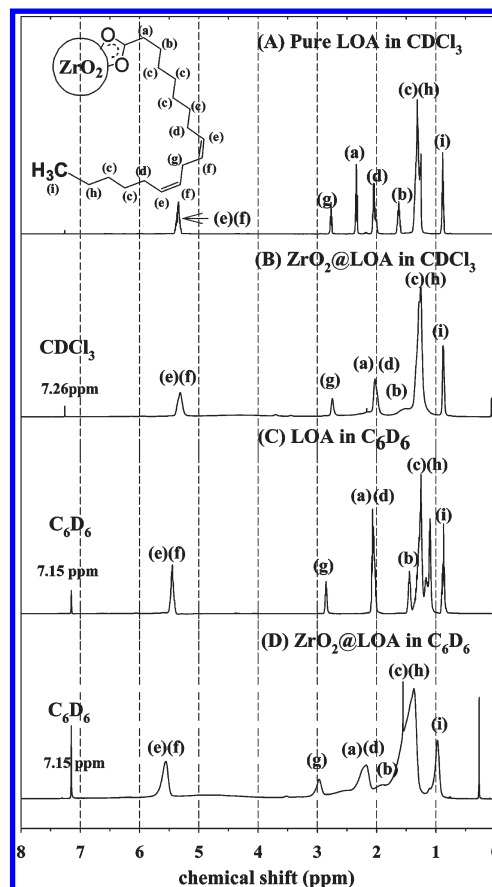
findings. A different proposition is thus needed to explain the shrinking of the shell thickness in chloroform.

Given that linoleic acid is an unsaturated fatty acid containing two cis double bonds, we speculate that these double bonds might interact with the aromatic ring of toluene and benzene. It would be easier for the aromatic solvent to swell and “interact” with the grafted linoleic acid, thus creating a brush conformation, as schematically illustrated in Figure 5. Without the intervention of solvent, the intrachain  $\pi$ - $\pi$  interaction of linoleic acid might induce a chain folding that leads to a thinner and more compact shell.

The interaction between LOA molecules with aromatic-ring-containing solvents was verified by liquid-state 500 MHz <sup>1</sup>H NMR measurements. Dispersions (10 wt %) of LOA-ZrO<sub>2</sub>, as for the SANS measurements, and pure LOA were prepared in deuterated benzene (C<sub>6</sub>D<sub>6</sub>) and chloroform (CDCl<sub>3</sub>). The <sup>1</sup>H NMR spectra of the dissolved LOA and LOA-ZrO<sub>2</sub> in these deuterated solvents are shown in Figure 6, where the signals are assigned according to the <sup>1</sup>H NMR decoupling experiments (see the Supporting Information, Figure S6).

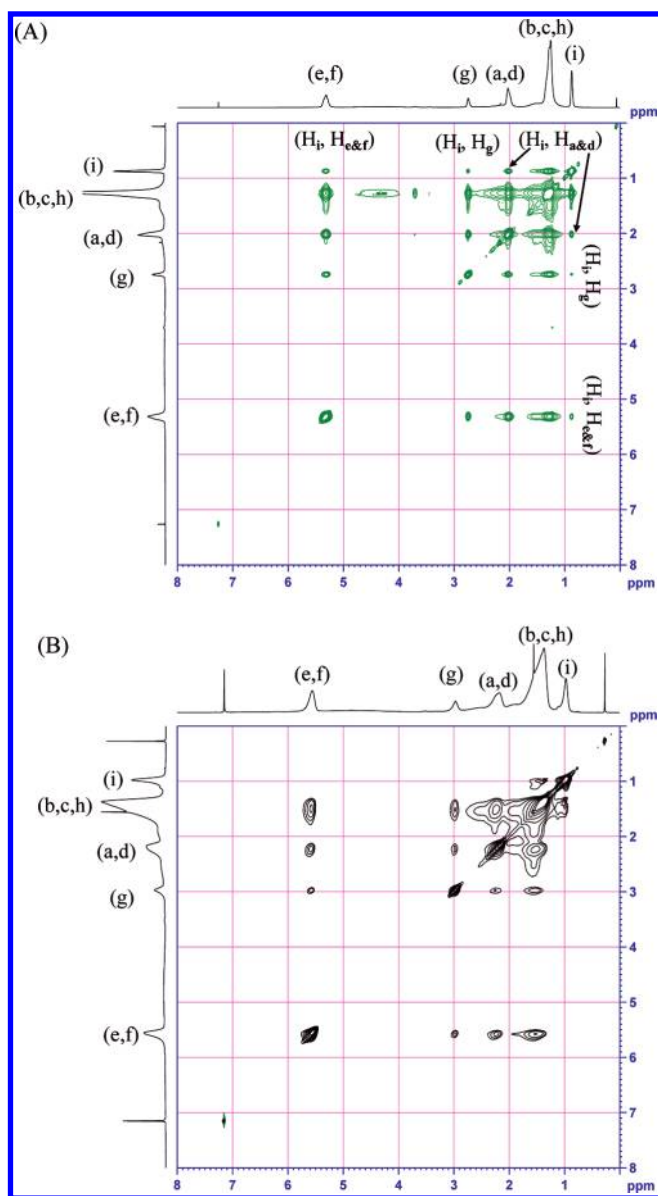
Figure 6 revealed different chemical shifts for the dissolved LOA molecules in CDCl<sub>3</sub> (Figure 6A) and in C<sub>6</sub>D<sub>6</sub> (Figure 6C). The differences are particularly obvious for the signals of CH<sub>2</sub> groups (H<sub>a&b</sub>) close to the carboxylic group and that of CH<sub>2</sub> groups (H<sub>d&g</sub>) in the vicinity of the alkene. The carboxylic group is an electron-withdrawing substituent. Therefore, the nucleus of CH<sub>2</sub> groups in the vicinity of the substituent is deshielded due to the reduced electron density. The electron-withdrawing ability of the carboxylic group is affected by its interaction with the solvent and is weaker in C<sub>6</sub>D<sub>6</sub> than in CDCl<sub>3</sub>. Consequently, the deshielding effect is weaker in C<sub>6</sub>D<sub>6</sub> and the H<sub>a</sub> and H<sub>b</sub> appeared relatively upfield compared with that in CDCl<sub>3</sub>.<sup>34,35</sup> On the other hand, the signals for H<sub>e-g</sub> appeared approximately 0.1–0.135 ppm downfield in C<sub>6</sub>D<sub>6</sub> than that in CDCl<sub>3</sub>. The downfield shifts of H<sub>e-g</sub> is most likely the result of the induced deshielding from the interaction of the double bonds with the more polarizable  $\pi$  electrons of the C<sub>6</sub>D<sub>6</sub> phenyl rings.

The <sup>1</sup>H NMR spectra (Figure 6B,D) of the LOA-ZrO<sub>2</sub> demonstrate the effects of anchoring LOA on ZrO<sub>2</sub>. Particularly,



**Figure 6.** Proton NMR spectra of dissolved LOA and LOA-ZrO<sub>2</sub> in deuterated (A, B) chloroform and (C, D) benzene. The inset reveals the chemical structure with proton positions labeled.

the signals of H<sub>a-b</sub> almost disappeared and turned into two extremely broad peaks. This arose from the dramatically reduced mobility of the grafted LOA.<sup>36</sup> The solvent influence is also present here, as all signals for the grafted LOA moved downfield



**Figure 7.** 2D NOESY spectra of LOA–ZrO<sub>2</sub> dispersed in deuterated (A) chloroform and (B) benzene.

in C<sub>6</sub>D<sub>6</sub>. Solvent is known to exert a strong influence on the cation–anion ion pair complexes, where the cation–anion interaction is stronger in less polar solvent (e.g., C<sub>6</sub>D<sub>6</sub>).<sup>37</sup> In C<sub>6</sub>D<sub>6</sub>, the ionic interaction between the carboxylic group with the ZrO<sub>2</sub> surface may be stronger and affects the whole LOA chain.

The organic shell structure of the LOA–ZrO<sub>2</sub> was further investigated by 2D NOESY NMR. Figure 7 shows the 2D NOESY spectra of LOA–ZrO<sub>2</sub> dispersed in CDCl<sub>3</sub> and C<sub>6</sub>D<sub>6</sub>. The spectrum displays diagonal peaks and crosspeaks. The latter correspond to pairs of protons that are near neighbor in space (<5 Å). The intensities of the crosspeaks decrease with the sixth power of the distance between the protons.<sup>34</sup> The crosspeaks between H<sub>i</sub>–H<sub>e&f</sub>, H<sub>i</sub>–H<sub>g</sub>, and H<sub>i</sub>–H<sub>a&d</sub> pairs, which appear in the CDCl<sub>3</sub> case (Figure 7A), but not in the C<sub>6</sub>D<sub>6</sub> case (Figure 7B), are particularly interesting. The absence or much weaker intensity of these crosspeaks in the C<sub>6</sub>D<sub>6</sub> indicates that these pairs are separated by a longer distance in C<sub>6</sub>D<sub>6</sub>, as it should

if the grafted chains are swollen by the solvent and stretch into a brushlike conformation. On the other hand, the proximity between H<sub>i</sub> with H<sub>a&d</sub> strongly suggests that the grafted LOA chains are folded in CDCl<sub>3</sub>, which, therefore, confirms our interpretation based on the SANS results. However, the proposed folding of LOA on the particle surface might not be the only possible configuration that leads to an enhancement of H<sub>i</sub>–H<sub>g</sub> and H<sub>i</sub>–H<sub>e&f</sub> crosspeaks. The enhancement of these crosspeaks might be attributed to a lower intermolecular distance; namely, the swelling (expansion) and deswelling (collapse) of the shell by the solvent lead to a different orientation of the LOA molecules on the particle surface.<sup>38</sup> Alternatively a monolayer of LOA with their alkyl chains all tilted in the same direction might also lead to a thinner shell and shorten the distance between protons of different molecules. However, although this may be conceivable on a flat surface, such an orientation on the outer surface of a particle would be difficult to imagine. Further clarification will be carried out as future work.

## CONCLUSION

The size distributions of dispersed nanoparticles are routinely characterized with TEM or DLS. However, TEM gives only the structural details in the dry form, which may be different from the situation in liquid. DLS probes the system in the dispersed state, but the result is affected by the proper assignment of the refractive indexes for the particles and the solvent. When inorganic nanoparticles are small and modified by organics, the latter may occupy an appreciable volume fraction, thus making the assignment of its refractive index difficult. The uncertainty in the refractive index leads to a rough estimation of the particle size. Only with the SAXS/SANS measurements is one able to obtain precisely the structural details of surface-modified nanoparticles, including the sizes of the inorganic core and the organic shell, as well as its secondary clusters.

The clustering and dispersion behavior of carboxylic acid-modified zirconia nanoparticles in different solvents as determined from SAXS/SANS measurements has been presented. In most cases, the transparent colloidal sol consists of “soft” secondary clusters generated from the nanoparticles. Only in one special condition, with linoleic acid modification and in chloroform, were the nanoparticles dispersed as primary units. The fact that the same entity appeared as secondary clusters in toluene and benzene, but as primary nanoparticles in chloroform, gave us an opportunity to understand the effect of solvent on the stabilization of the colloidal system.

The findings indicate that a colloidal system can be stabilized as either primary nanoparticles or aggregated clusters depending on the conformation of the grafting organic layer. The layer exhibits a swollen brush conformation when there are strong interactions between the grafted molecules and the solvent. Cluster dispersion is favored under such a circumstance. Conversely, a more compact organic layer is formed if the solvent does not swell the grafted molecules. The dispersion can then be stabilized as primary particles.

## ASSOCIATED CONTENT

**S Supporting Information.** FTIR spectra of carboxylic acid-modified ZrO<sub>2</sub> (Figure S1); TGA of carboxylic acid-modified ZrO<sub>2</sub> (Figure S2); UV/vis spectra of LOA-modified ZrO<sub>2</sub> in different solvents (Figure S3); size distributions of LOA–ZrO<sub>2</sub>



dispersed in chloroform, benzene, and toluene measured by DLS (Figure S4); SAXS data and corresponding Kratky plot of LOA–ZrO<sub>2</sub> dispersed in benzene and toluene (Figure S5); and 1D NMR spectrum and decoupled spectra of LOA–ZrO<sub>2</sub> dispersed in CDCl<sub>3</sub> (Figure S6). This material is available free of charge via the Internet at <http://pubs.acs.org>.

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