Article

pubs.acs.org/JPCC

Fast One-Step Synthesis of Biocompatible ZnO/Au Nanocomposites with Hollow Doughnut-Like and Other Controlled Morphologies

- 3 Jun Geng,^{†,‡} Guang-Hui Song,[†] Xiang-Dong Jia,[†] Fang-Fang Cheng,[†] and Jun-Jie Zhu*,[†]
- 4 †State Key Laboratory of Analytical Chemistry for Life Science, School of Chemistry and Chemical Engineering, Nanjing University,
- 5 Nanjing 210093, P. R. China

8

9

10

11

12

13

14

15

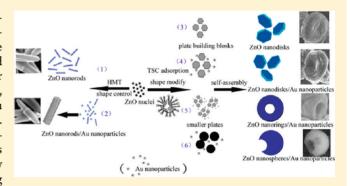
16 17

18

19

6 [‡]Department of Chemistry, Jiangsu Institute of Education, Nanjing 210013, P. R. China

ABSTRACT: Hollow doughnut-like ZnO/Au nanocomposties have been synthesized through a fast one-step microwave-assisted hydrothermal route. The formation mechanism of the product is closely connected with the coordination and selective adsorption effect of trisodium citrate (TSC). Other different structures, such as ZnO nanorods/Au nanoparticles, ZnO nanodisks/Au nanoparticles, and ZnO nanospheres/Au nanoparticles have also been obtained in control experiments. The UV—vis and photoluminescence properties of the asprepared hollow structures with different Au concentrations showed tunable UV and visible emission intensity. The hollow ZnO/Au nanocomposites are photostable with a strong resonance Raman signal. The colorimetric 3-(4,5-dimethylth-



iazol-2-yl)-2,5-diphenyl tetrazolium bromide (MTT) assays showed that ZnO/Au nanomaterials have low biological cytotoxicity on human colon cancer cells (LOVO cells) at the concentration of 50 μ g/mL. The as-prepared ZnO/Au nanocomposites with good biocompatibility may have potential applications in biomedicine and biosensors.

1. INTRODUCTION

23 The optical, electronic, and magnetic properties of semi-24 conductor and metal nanoparticles have been investigated 25 intensively, and their functionality has been improved by 26 forming hybrid nanocrystals. 1-3 Particularly, the presence of 27 metal nanoparticles on the surface of semiconductor nano-28 structures was expected to change the optical and electronic 29 properties due to the couplings of metal and semiconductor. 4,5 30 The noble metal nanoparticles possess many unique properties 31 such as easy reduction, high chemical stability, and $_{32}$ bioaffinity; $^{6-8}$ in addition, they can enhance the activity of 33 photocatalyst due to good charge storage capability. 9 As the 34 noble metal nanoparticles are combined with the semi-35 conductor nanocrystals, the as-formed hybrid nanocrystals 36 exhibit photoluminescence (PL) properties superior to that of 37 pure semiconductor nanocrystals. 10 Coupling of an oxide 38 semiconductor and noble metal also allows tailoring of surface 39 plasmon resonance (SPR) by electron transfer for plasmon-40 based chemical and biological sensors. 11,12

As an important II—VI semiconductor with a wide bandgap (3.37 eV) and a large exciton binding energy (60 meV), ZnO 43 has received much attention over the past decade due to its 44 unique optical, acoustic, and electronic properties. 13,14 Mean-45 while, they are more environmentally friendly than those Cd-46 containing semiconductor nanocrystals. Meanwhile, Au nanoparticles have attracted considerable attention because of 48 their remarkable catalytic and optical properties and numerous 49 potential applications in surface-enhanced Raman scattering

and chemical and biological sensing. ^{16–18} Au is explored to 50 combine ZnO QDs to obtain biocompatible and water-soluble 51 material. The hybrid nanostructures containing Au and ZnO 52 are the attractive nanocomposites due to their novel or 53 enhanced performances, such as very promising nonlinear 54 optical properties, ¹⁹ efficient fluorescence resonance energy 55 transfer properties, ²⁰ and enhanced sensitivity in detecting CO 56 gas. ²¹ The electric field enhancement in biocompatible ZnO/ 57 Au nanocomposites can generate strong resonance Raman 58 signals that can be used for ultrasensitive DNA²² and protein ²³ 59 detection.

Although there are some reports concerning the synthesis of 61 ZnO/Au nanocomposites, the morphology of final products is 62 only confined to ZnO nanoparticles/Au nanoparticles^{11,22} and 63 ZnO nanorods/Au nanoparticles.^{24–26} No other morphology 64 has been reported. Besides, it was found to be difficult to 65 control the size and morphology of the hybrid nanocryst-66 als.^{11,27,28} In the present work, a ZnO/Au nanocomposite with 67 special hollow doughnut-like structure was fabricated via a fast 68 one-step microwave-assisted hydrothermal method. The 69 reaction conditions were investigated to find the key factor in 70 the formation of this hollow structure. Other structures, such as 71 ZnO nanorods/Au nanoparticles, hexagonal ZnO nanodisks/ 72 Au nanoparticles, and ZnO nanospheres/Au nanoparticles 73

Received: December 15, 2011 Revised: January 20, 2012



74 could also be obtained by controlling experimental conditions.
75 The UV—vis absorption and photoluminescence spectra were
76 measured to show the interaction of the ZnO framework and
77 Au nanoparticles. Strong resonance Raman spectroscopic
78 signals under the excitation of an ultraviolet laser could be
79 obtained for the as-prepared ZnO/Au nanocomposites. The
80 cytotoxicity of the nanocomposites has been evaluated. To our
81 best knowledge, controlled synthesis of hollow ZnO/Au
82 nanocomposites with enhanced resonance Raman signal and
83 tunable visible emission intensity has not been reported so far.

2. EXPERIMENTAL SECTION

2.1. Materials. All the reagents used were of analytical spurity and were used without further purification. Zn-86 (NO₃)₂·6H₂O and hexamethylenetetramine (HMT) were purchased from Beijing Chemical Reagents Company Limited sof China. Chloroauric acid (HAuCl₄) and trisodium citrate (TSC) were obtained from Shanghai Reagent Company of China. 3-(4,5-Dimethylthiazol-2-yl)-2,5-diphenyl tetrazolium bromide (MTT) was purchased from AMRESCO, and dimethylsulfoxide (DMSO) was obtained from Sinopharm Chemical Reagent Company Limited of China.

2.2. Apparatus. A microwave synthetic system (CEM ps Discover, USA) was used for the preparation of ZnO/Au nanocrystals, which was equipped with controllable temperature and pressure units. The system can operate at 2450 MHz ps frequency and work at 0–300 W power. The reaction ps temperature, pressure, and time can be programmed. The syntheses of nanocrystals were preformed in a cylindrical digestion vessel that was a high-strength vessel consisting of a pecial kind of glass. The volume of the vessel used in the reaction was 80 mL.

The X-ray powder diffraction (XRD) analysis was performed 105 on a Japan Shimadzu XRD-6000 powder X-ray diffractometer 106 at a scanning rate of 4° /min in the 2θ range from 10° to 80° , 107 with graphite-monochromatized Cu K α radiation ($\lambda = 0.15418$ 108 nm). The scanning electron micrographs (SEM) were taken on 109 an LEO-1530VP field-emission scanning electron microscope. 110 Transmission electron microscopy (TEM) was carried out on a 111 HITACHI H-7650 transmission electron microscope, using an 112 accelerating voltage of 200 kV. High-resolution transmission 113 electron micrographs (HRTEM), selected area electron 114 diffraction (SAED) patterns, and energy dispersive X-ray (EDX) analysis were obtained by employing a JEOL-2010 116 high-resolution transmission electron microscope with an accelerating voltage of 200 kV. UV-vis absorption spectra 117 were obtained using a Shimadzu UV-3600 spectrophotometer. 119 Photoluminescence spectra (PL) were measured on a 120 Shimadzu RF-5301PC fluorescence spectrometer at room 121 temperature. Resonanace Raman scattering spectra were 122 taken on a Labram HR800 spectrophotometer (Jobin Yvon, France) with the excitation wavelength of 325.0 nm.

2.3. Synthesis. Typically, Zn(NO₃)₂·6H₂O (0.0025 mol) was dissolved in 25 mL of H₂O, and then HMT (0.0025 mol) was introduced with stirring to form a clear solution A. An 127 amount of 1 mL of 1% HAuCl₄ was introduced to the solution 128 containing 15 mL of 5 mmol·L⁻¹ TSC to make solution B. The mixture of solution A and B was transferred to an 80 mL 130 cylindrical digestion vessel under agitation. The reaction was 131 maintained at 100 °C and 175 psi for 30 min under microwave 132 irradiation (260 W). A purple precipitate was centrifuged, 133 washed with distilled water and absolute ethanol in sequence,

and finally dried in air. The final products were collected for 134 characterization.

2.4. Cell Cytotoxicity Measurement. Human colon 136 cancer cells (LOVO cells) were plated in a 96-well plate at 1 137 × 10⁴ cells/well. After 24 h incubation, the medium was 138 replaced with 100 µL of fresh medium containing the ZnO/Au 139 nanocomposites at different concentrations (50, 25, 10, 5, 1, 140 and 0 μ g·mL⁻¹). Cells were kept at 37 °C for 24 and 48 h. 141 Then the medium was removed, and 100 μ L of fresh serum- 142 free medium containing 20 µL of colorimetric 3-(4,5-143 dimethylthiazol-2-yl)-2,5-diphenyl tetrazolium bromide 144 (MTT, 5 mg/mL, in PBS) was added into each well. After 4 145 h of incubation, MTT was converted into an insoluble purple 146 formazan dye by mitochondrial dehydrogenases of living cells. 147 The media were removed, and cells were lysed with dimethyl 148 sulfoxide (DMSO). Absorbance was measured at 490 nm using 149 a microplate reader (Bio-Rad 680, USA) to assess the relative 150 viability of the cells. The following formula was used to 151 calculate the inhibition of cell growth

Cell viability (%) = (mean absorbance value of the treatment group/mean absorbance value of control) \times 100%

3. RESULTS AND DISCUSSION

3.1. Characterization of the Final Products. Figure 1 153 ft shows the XRD pattern of the as-prepared product. A 154

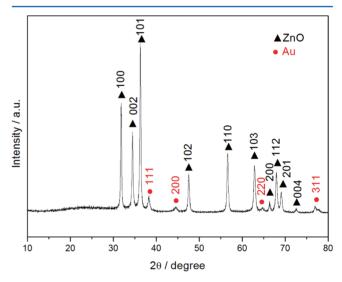


Figure 1. XRD pattern of the as-prepared hollow ZnO/Au nanocomposites.

hexagonal phase ZnO structure (space group $P6_3mc$) is found $_{155}$ coexistent with cubic phase Au (space group Fm_3m) in the $_{156}$ XRD pattern of the composite structures. All of the diffraction $_{157}$ peaks in the XRD pattern are well assigned to ZnO and Au as $_{158}$ reported in JCPDS card No. 36-1451 and 04-0784, respectively. $_{159}$

The morphologies of ZnO/Au nanocomposites were $_{160}$ observed by SEM and TEM images. Figure $_{2a-c}$ shows the $_{161\ f2}$ typical SEM images of the as-prepared hollow ZnO/Au $_{162}$ nanocrystals. It is clearly demonstrated that the crystals have $_{163}$ a special hollow doughnut-like structure with an outer diameter $_{164}$ of 800 nm and an inner diameter of about 250 nm. The shell of $_{165}$ the hollow structure seems to be composed of flake-like $_{166}$

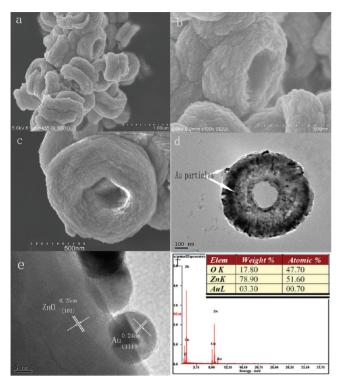


Figure 2. (a) Low-magnification and (b,c) high-magnification SEM images, (d) TEM image, (e) HRTEM image, and (f) EDX spectrum of the as-prepared hollow doughnut-like ZnO/Au nanocomposites with a ratio of Au at 1%.

167 building blocks. The morphology and microstructure of the as-168 synthesized ZnO/Au hollow crystals were further studied by 169 TEM and HRTEM. Figure 2d shows that the as-prepared 170 ZnO/Au sample appears as hollow nanodoughnuts with diameter of ca. 800 nm. A strong contrast difference between 172 the edges (dark) and centers (bright) indicates hollow interiors with a wall thickness of about 250-300 nm. Small dark Au nanoparticles with size of about 10 nm could be detected (Figure 2d). In our synthetic route, HAuCl₄ was reduced at the same time when ZnO was formed. Therefore, Au nanoparticles were located randomly in the framework or on the surface of ZnO. The HRTEM image obtained on the surface of a nanoring (Figure 2e) indicates clear interplanar spacings of 0.25 180 nm, corresponding to the (101) crystal faces of ZnO. The 181 HRTEM image recorded on the small dark nanoparticles shows 182 interplanar spacings of 0.24 nm, which could be indexed to the (111) crystal faces of cubic phase Au. To further confirm the composition of the product, EDX spectra were recorded. The EDX spectrum acquired from one doughnut-like structure (Figure 2f) reveals the presence of Zn, O, and Au with an atomic percentage of 51.60%, 47.70%, and 00.70%, which is with an approximate atomic ratio of 1.000:0.924:0.014. This 189 spectrum indicates that the as-prepared product is ZnO/Au 190 nanocomposites with a ratio of Au at about 1%.

3.2. Possible Formation Mechanism. Control experiments were conducted to investigate the possible formation mechanism of the special hollow doughnut-like nanostructure. The relationship between different reaction systems and final ps products was summarized in Table 1 and Figure 3.

When reactants only concluded Zn²⁺ and HMT, pure ZnO lipstick-like nanorods with length of about 600 nm and lipst diameter of about 100 nm were obtained, as shown in Figure 4a

Table 1. Compositions and Morphologies of Products Prepared in Different Reaction Systems

systems	reactants	compositions	morphologies
(1)	$Zn^{2+} + HMT$	ZnO	ZnO nanorods
(2)	$Zn^{2+} + HMT + HAuCl_4$	ZnO/Au	ZnO nanorods/Au nanoparticles
(3)	$Zn^{2+} + HMT + TSC$	ZnO	ZnO nanodisks
(4)	Zn ²⁺ + HMT + HAuCl ₄ + 3 mM TSC	ZnO/Au	ZnO nanodisks/Au nanoparticles
(5)	Zn ²⁺ + HMT + HAuCl ₄ + 5 mM TSC	ZnO/Au	ZnO nanorings/Au nanoparticles (hollow doughnut-like nanostructure)
(6)	Zn ²⁺ +HMT +HAuCl ₄ +10 mM TSC	ZnO/Au	ZnO nanospheres/Au nanoparticles

and b. SAED measurement showed that these nanorods grew 199 preferentially along the c-axis ([001] direction). In this case, 200 HMT is a nontoxic, water-soluble, nonionic cyclic tertiary 201 amine, which can be reversibly converted into ammonia (NH $_3$) 202 and formaldehyde (HCHO) through heat treatment (eq 1). 203 In the presence of ammonia, the $[\mathrm{Zn}(\mathrm{NH}_3)_4]^{2+}$ complex can be 204 formed consequently (eq 2). After that, $[\mathrm{Zn}(\mathrm{NH}_3)_4]^{2+}$ 205 transforms to zinc oxide during the crystallization. 206

$$(CH_2)_6N_4 + 6H_2O \rightarrow 4NH_3 + 6HCHO$$
 (1)

$$Zn^{2+} + 4NH_3 + 4H_2O$$

 $\rightarrow Zn(NH_3)_4^{2+} + 4H_2O$
 $\rightarrow ZnO + 2OH^- + 4NH_4^+ + H_2O$ (2)

Additionally, HMT can kinetically control species in solution 207 by coordinating to Zn^{2+} and keeping the free zinc ion with low 208 concentration. HMT can also coordinate to the ZnO crystal, 209 hindering the growth of certain surfaces. With additional 210 systematic studies, one or all of these mechanisms may be 211 found to be responsible for the HMT shape control. Here, 212 HMT acts as a pH buffer by slowly decomposing to provide a 213 gradual and controlled supply of ammonia, which can form 214 ammonium hydroxide as well as complex zinc(II) to form 215 $\left[Zn(NH_3)_4\right]^{2+\frac{31}{32}}$ Because dehydration of the zinc hydroxide 216 intermediates controls the growth of ZnO, the slow release of 217 hydroxide may have a profound effect on the kinetics of the 218 reaction. A similar shape-controlled effect of HMT for ZnO 219 nanorods has been reported. 33,34

While 1 mL of 1% HAuCl₄ was added in the above reaction 221 system of Zn²⁺ and HMT, ZnO nanorod/Au nanoparticle 222 composites were synthesized as shown in Figure 4c and d. This 223 result showed that HAuCl₄ could be reduced by HMT, and the 224 product remained the rodlike morphology because of the 225 shape-control effect of HMT. Thus, ZnO/Au composites can 226 be obtained in a one-pot process utilizing the ammonia and 227 formaldehyde released by HMT during heat treatment.

In another control experiment, 15 mL of 3 mM TSC was 229 added into the reaction system with $\rm Zn^{2+}$ and HMT but 230 without HAuCl₄, and pure ZnO hexagonal nanodisks with 231 diameter of 700 nm and thickness of about 200 nm were 232 prepared (Figure 4e and f). SAED result (inset of Figure 4f) 233 revealed that the nanodisks were perpendicular to the c axis. 234 The transformation from nanorods to nanodisks indicated that 235 TSC could adjust the morphology, and its effects on shape 236 control surpassed HMT. As we known, ZnO normally has a 237

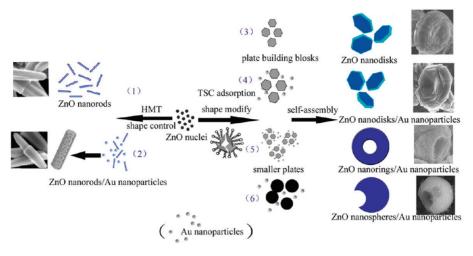


Figure 3. Schematic illustration of the formation of ZnO and ZnO/Au nanostructures.

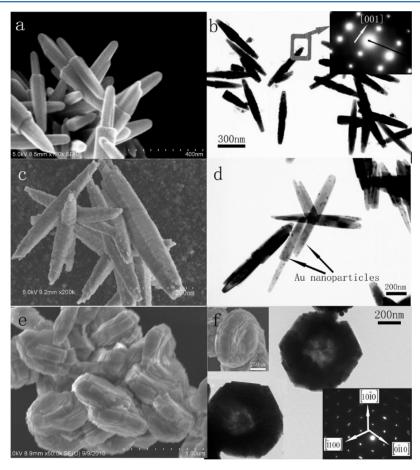


Figure 4. SEM and TEM images of products obtained in different reaction systems: (a, b) ZnO nanorods obtained in the reaction system of Zn^{2+} and HMT, (c, d) ZnO nanorods/Au nanoparticles obtained in the reaction system of Zn^{2+} , HMT, and HAuCl₄, and (e, f) ZnO nanodisks obtained in the reaction system of Zn^{2+} , HMT, and TSC. The insets of b and f are the SAED patterns of each product, respectively.

238 rodlike morphology. The growth of oriented nanodisks is 239 unusual and is directly related to the function of organic acid 240 (citrate), which inhibits the longitudinal growth of rods but 241 promotes the growth of plates. The citrate anion is an 242 important biological ligand which can coordinate with metal 243 ions such as Ag⁺, Co²⁺, Cu²⁺, Fe³⁺, Fe²⁺, Zn²⁺, and Ni²⁺ to form 244 strong complexes. 35 Besides, it is also a shape modifier. 36 Citrate anions have been known to act as a capping agent of the 246 (0001) surface of the ZnO crystal by adsorbing on the positive

polar face of the (0001) surface. 36,37 When citrate anions are 247 adsorbed on the (0001) surface, these citrate anions prevent 248 the contact between $\rm Zn(OH)_4^{2^-}$ and the (0001) ZnO crystal 249 surface. As the growth along the [0001] direction is suppressed 250 by citrate anions, ZnO crystal growth will mainly proceed along 251 the six symmetric directions to form hexagonal disks.

In the reaction system containing Zn²⁺, HMT, HAuCl₄, and 253 TSC (15 mL, 5 mM), ZnO nanorings/Au nanoparticles were 254 prepared (Figure 2). Small flake-like building blocks could be 255

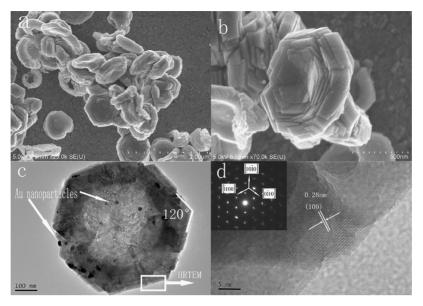


Figure 5. (a) Low-magnification and (b) high-magnification SEM images, (c) TEM image, and (d) HRTEM image of the ZnO nanodisk/Au nanoparticle composites. The inset of d is the SAED pattern of the product.

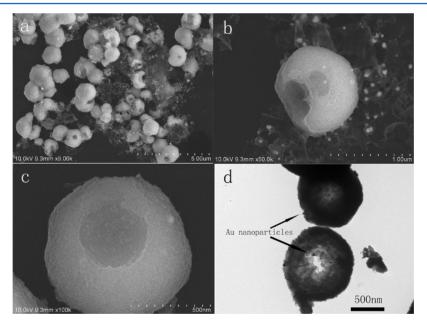


Figure 6. (a) Low-magnification SEM image, (b, c) typical high-magnification SEM images from different angles, and (d) TEM image of the hollow and semihollow spherical ZnO/Au nanocomposites obtained with 10 mM TSC.

256 detected in the nanostructure. In the growth of ZnO/Au 257 hollow doughnut-like architectures, citrate anions play at least 258 three roles. First, TSC could reduce chloroauric acid to Au 259 nanoparticles. Second, citrate anions coordinate with Zn²⁺ to 260 form citrate—zinc complexes, which prevent the high free zinc 261 ion concentration. The formation of citrate—zinc complexes also decreases the generation rate of ZnO nanoparticles 263 in solution. The relatively slow generation rate of ZnO 264 nanoparticles is favorable for the subsequent growth of 265 nanoplates along the determinate direction and the self-266 assembly to hollow nanorings. Finally, the citrate anion can 267 also serve as a shape modifier which selectively binds to some 268 crystal planes and restricts their growth, whose effect was also 269 applied in the synthesis of ZnO 3D architectures. The 270 malate anion, which has a similar molecular structure, plays the 271 same role in the fabrication of ZnO hexagonal architectures.

The interaction between citrate anions and nanoparticles would 272 also facilitate crystals to grow along the determinate direction. 273 Besides, there is an interaction between citrate anions binding 274 to the nanoparticles, such as van der Waals forces and 275 intermolecular hydrogen bonds. The interaction would induce 276 the assembly of nanoparticles forming ZnO nanorings and 277 landing Au nanoparticles on the skeleton of ZnO, which further 278 resulted in the formation of ZnO/Au doughnut-like architec- 279 tures.

In this synthesis, microwave irradiation was fast and highly 281 efficient for transferring energy into the reaction system, and 282 the temperature increased uniformly throughout the reac- 283 tants. High temperature (100 °C) to the advantage of 284 reducing Au on the ZnO nanocrystals could be easily obtained 285 in 5 min, and a fast and homogeneous nucleation process could 286 be achieved, which improves the crystallinity of the ZnO/Au 287

288 nanocrystals. In the procedure, the time required to attain good 289 crystallinity and uniform size of ZnO/Au nanocrystals was 290 within half an hour. We have tried control experiments with a 291 shortened reaction time of 10 min, and the same ZnO/Au 292 doughnut-like nanostructure could also be obtained.

3.3. Effect of TSC Concentrations. In the reaction system containing Zn²⁺, HMT, HAuCl₄, and TSC, ZnO nanorings/Au 295 nanoparticles, ZnO nanodisks/Au nanoparticles, or ZnO 296 nanospheres/Au nanoparticles could be prepared according 297 to the concentration of TSC.

When the concentration of TSC was 3 mM, the ZnO nanodisk/Au nanoparticle composite was obtained. The typical SEM image showed that the hexgonal dished nanodisks with diameter of about 600 nm were composed of many hexgonal sheetlike building blocks (Figure 5a and b). TEM image showed that small Au nanoparticles were attached on the hexagonal disks. The angle between two adjacent edges of one disk was measured to be 120°. The SAED pattern recorded on the edge of the structures shows that the ZnO framework is of single-crystal nature with hexagonal phase (inset of Figure 5d). The HRTEM image recorded on the edge of nanodisks shows clear interplanar spacings of 0.28 nm, which can be indexed to the (100) crystal faces of hexagonal phase ZnO. Combined with the SAED measurement, we could draw the conclusion that the nanodisks were perpendicular to the c axis.

After increasing TSC concentration to 5 mM, the ZnO 314 nanoring/Au nanoparticle composite was fabricated, as shown 315 in Figure 2. By careful observation, it is found that the 316 nanorings were composed of sheet-like crystals growing layer 317 by layer. The size of these sheet-like building blocks was smaller 318 than that of building blocks in nanodisks prepared with 3 mM 319 TSC. The increase of TSC concentration might cause more 320 citrate-zinc complexes to form and more citrate anions to selectively bind to crystal planes, which therefore restricted 322 their crystal growth and resulted in smaller plates. The increase of TSC also had an effect on the transformation from nanodisks 324 to hollow nanorings. We supposed that the organic acid citrate 325 with suitable concentration might be responsible for the 326 formation of nanorings. However, the exact function of this 327 TSC concentration is not very clear and still under 328 investigation. We think that the synergistic action of TSC, 329 HMT, and Zn²⁺ might be responsible for the formation of 330 hollow nanorings.

Further increasing TSC concentration to 10 mM resulted in hollow and semihollow spherical ZnO/Au nanocomposites, as shown in Figure 6. Some free small Au nanoparticles could also seen. The disk-like building blocks could not be detected in these spheres, which was different from the above hollow nanorings. The excess amount of TSC might cause extra adsorption and restriction to nanoparticles growth, which resulted in the disappearance of hexagonal disk-like building blocks. More TSC also caused faster reduction of HAuCl₄, and therefore more Au nanoparticles did not land on the framework of ZnO but existed as free Au nanoparticles.

342 **3.4. Optical Properties Measurement.** ZnO is a wide-343 band-gap semiconductor ($E_{\rm g}=3.37~{\rm eV}$) with a large exciton 344 binding energy (60 meV). Optical studies were performed to 345 evaluate the potentially optical qualities of the doughnut-shaped 346 ZnO/Au nanocomposites.

Figure 7 showed the UV—vis absorption spectra of pure ZnO nanorods and ZnO nanoring/Au nanoparticle composites with different Au concentrations. The absorption peaked at 370 nm shows exhibits a well-defined exciton band of ZnO nanorods (Figure

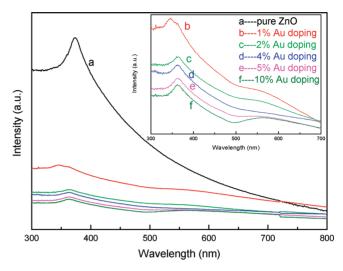


Figure 7. Absorption spectra of (a) pure ZnO and ZnO/Au nanocomposites with a ratio of Au at (b) 1%, (c) 2%, (d) 4%, (e) 5%, and (f) 10%. The inset shows the amplificatory absorption spectra of the ZnO/Au nanocomposites.

7a). The calculated band gap is 3.35 eV under current 351 measured conditions, which is in accordance with the bulk 352 values. No quantum confinement effects were observed because 353 of the relatively large size.

The exciton band of ZnO/Au nanocomposites showed a 355 blue-shift to around 360 nm, which may result from the small 356 nanodisk building blocks in the nanocomposites as discussed 357 above. In the inset of Figure 7, it is clear that the plasmon 358 absorption band centered at 550 nm of Au appears with the 359 formation of ZnO/Au nanocomposites. This peak has a red-360 shift from 540 to 560 nm with the increased amount of Au on 361 the surface of ZnO from 1 to 10%. It can be attributed to the 362 electron transfer between Au and ZnO, which provides support 363 to the formation of the ZnO/Au nanocomposite. 364

The optical properties of ZnO/Au nanocrystals were further 365 manifested by the room-temperature photoluminescence (PL) 366 spectra under photon excitation of 325 nm. It can be observed 367 that two emission bands appeared in the PL spectra of pure 368 ZnO (Figure 8a). The emission band centered at 363 nm could 369 f8 be attributed to the transition between the band edges, and the 370 band centered at 525 nm in the yellow region could be 371 attributed to the intrinsic defects of oxygen vacancies related 372 emission. 42 As shown in Figure 8b-f, ZnO/Au hybrid 373 nanostructures with different Au concentrations all exhibit 374 stronger UV emission around 360 nm and weaker visible 375 emission at about 500 nm than those of pure ZnO. For ZnO/ 376 Au hybrid nanocrystals, upon increasing the amount of Au, the 377 UV emission intensity is enhanced, and the visible emission is 378 suppressed gradually.

There are two possible reasons for the enhancement of UV 380 emission intensity and the decrease of visible emission intensity 381 in the ZnO/Au system: 43,44 (i) the adsorption of visible light by 382 Au can create exciton—surface plasmon coupling, which favors 383 the electron transfer from Au to ZnO, resulting in the 384 enhancement of UV emission intensity, and (ii) the number 385 of single ionized oxygen vacancies (Vo⁺) has been decreased/ 386 quenched by capturing electrons when more electrons 387 transferred from Au to ZnO. The surface defects of ZnO 388 become fewer and fewer by increasing the amount of Au, and 389 thus the visible emission becomes weaker and weaker in the 390



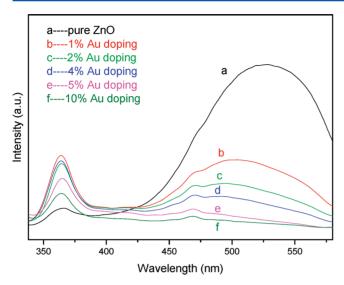


Figure 8. PL spectra of (a) the pure ZnO and ZnO/Au nanocomposites with ratio of Au at (b) 1%, (c) 2%, (d) 4%, (e) 5%, and (f) 10%.

391 present case, which is similar to ZnO/Ag heterostructures. 43 392 Even though these two explanations can explain well the 393 bleaching of the visible emission and enhancement of UV 394 emission intensity, the detailed mechanism needs more 395 dedicated work in the future.

396 **3.5. Resonance Raman Scattering Spectra.** ZnO 397 quantum dots (QDs) have the characteristic resonance 398 multiple-phonon Raman lines with the excitation wavelength 399 of 325 nm, which can be used as a characteristic fingerprint 400 signal. Upon forming ZnO/Au nanocomposites, the electrons 401 transferred between ZnO and Au nanoparticles can yield the 402 electromagnetic-field enhancement, which further enhances the 403 multiple-phonon Raman scattering of ZnO.¹¹

Figure 9 showed the 2D and 3D resonance Raman scattering spectra of pure ZnO and ZnO/Au nanocomposites. In Figure 406 9a, pure ZnO nanocrystals had a weak resonance Raman signal. The resonance Raman signal of ZnO capped by Au nanoparticles increased as shown in Figure 9b–f. Because the electron transfer between ZnO and Au nanoparticles occurred,

the built-in electrical field formed in the interface of ZnO/Au 410 nanocomposites. As a result, ZnO received the stronger local 411 electromagnetic field. The formed local electromagnetic field 412 can interact with optical phonons in ZnO nanocrystals, which 413 enhanced the scattering intensity of active phonons. 11 As 414 shown in Figure 9b-d, with increasing the amount of Au on 415 the surface of ZnO from 1 to 4%, the Raman signal increased. 416 The enhanced intensity indicated more electrons transfer 417 between Au and ZnO and therefore increased the electro- 418 magnetic field around ZnO nanocrystals. The electromagnetic 419 field enhancement further enhanced the multiple-phonon 420 Raman scattering of ZnO. However, when the amount of Au 421 on the surface of ZnO was greater than 4%, the Raman 422 intensity decreased (Figure 9e and f). Therefore, the amount of 423 Au was chosen as 4% to obtain an optimal Raman 424 enhancement. The ZnO/Au nanocomposites can serve as a 425 universal biocompatible tag for analyzing various biological 426 macromolecules such as proteins and DNA.

3.6. Cytotoxicity of ZnO/Au Nanocrystals for the 428 LOVO Cell Lines. Colorimetric 3-(4,5-dimethylthiazol-2-yl)- 429 2,5-diphenyl tetrazolium bromide (MTT) assays were 430 performed to estimate the cytotoxicity of the nanocomposites. 431 The MTT assays of the cells showed that ZnO/Au nanoma- 432 terials had low cytotoxicity at a concentration of 50 µg/mL on 433 LOVO cell lines (Figure 10) which indicated good biocompat- 434 f10 ibility and potential applications in biomedicine and biosensor. 435 It is found that the cytotoxicity of ZnO/Au nanocomposites 436 mainly comes from ZnO.²³ Also of note, there was a small 437 increase of cell viability for the ZnO/Au nanocrystals with 24 h 438 incubation, which might arise from the stimulation of cells in 439 the presence of nanoparticles.⁴⁵ Other Cd-containing semi- 440 conductor nanocrystals, such as CdSe, 15 CdTe, 46 and CdS, 47 441 are more toxic than ZnO. For example, CdTe QDs of low 442 concentrations (0.75 μ M) led to a significant death rate of 443 human erythroleukemia cells (K562 cells) after 24 or 48 h 444 incubation, 46 and these nanocrystals must be coated with some 445 biocompatible shells for biological applications. 15,48

4. CONCLUSIONS

In summary, hollow doughnut-shaped ZnO/Au nanocompo- 447 sites have been synthesized via a fast one-step microwave- 448

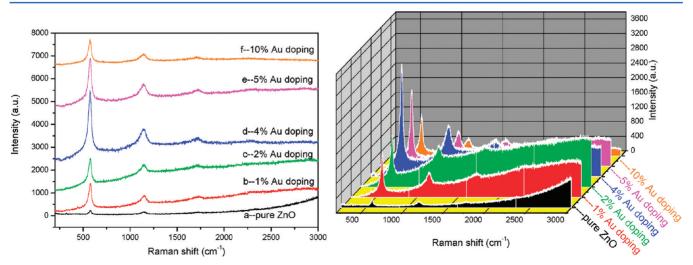


Figure 9. 2D (left) and 3D (right) resonance Raman scattering spectra of (a) pure ZnO and ZnO/Au nanocomposites with ratio of Au at (b) 1, (c) 2, (d) 4, (e) 5, and (f) 10% (λ_{exc} = 325 nm).

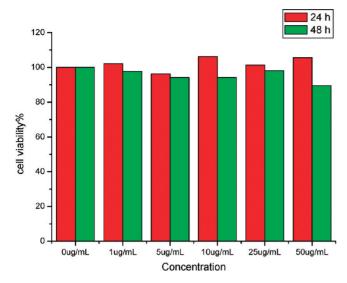


Figure 10. Cell viability of LOVO cells treated with different concentrations of hollow doughnut-shaped ZnO/Au nanocomposites for 24 and 48 h as measured by MTT assays.

449 assisted hydrothermal route. Citrate salts have been introduced 450 as shape modifiers and proved to be efficient to control the 451 shape of the ZnO/Au nanostructures. Control experiments 452 could also obtain other morphologies which enriched the 453 shape-controlled synthesis of nanocomposites. The PL spectra 454 of ZnO/Au nanocomposites exhibited stronger UV emission 455 and weaker visible emission upon increasing the amount of Au. 456 Compared with pure ZnO, the as-prepared ZnO/Au hybrid 457 crystals showed enhanced multiple-phonon Raman scattering, 458 which could serve as a universal biocompatible tag for analyzing 459 various biological macromolecules such as proteins and DNA. 460 The biological toxicity of the ZnO/Au nanocomposites has 461 been evaluated, and results showed good biocompatibility, 462 which promised a potential application in biomedicine and 463 biosensor.

464 AUTHOR INFORMATION

65 Corresponding Author

466 *E-mail: jjzhu@nju.edu.cn. Tel. and Fax: +86-25-83597204.

467 Notes

468 The authors declare no competing financial interest.

69 ACKNOWLEDGMENTS

470 This work is supported by National Natural Science 471 Foundation of China (Grant Nos. 20805022, 21121091) and 472 China Postdoctoral Science Foundation (Grant No. 473 20100471294). We also thank the support from Doctoral 474 Foundation from Ministry of Education (No. 475 20100091110023). We thank Ya-Jing Yin from Nanjing Normal 476 University for her kind help in TEM and resonance Raman 477 scattering measurements.

478 REFERENCES

- 479 (1) Kim, H.; Achermann, M.; Balet, L. P.; Hollingsworth, J. A.; 480 Klimov, V. I. *J. Am. Chem. Soc.* **2005**, *127*, 544.
- 481 (2) Lee, J.; Govorov, A. O.; Dulka, J.; Kotov, N. A. *Nano Lett.* **2004**, 482 *4*, 2323.
- 483 (3) Jian, D. L.; Gao, Q. M. Chem. Eng. J. 2006, 121, 9.
- 484 (4) Zhang, W.; Govorov, A. O.; Bryant, G. W. Phys. Rev. Lett. 2006, 485 97, 146804.
- (5) Lee, J.; Kotov, N. A.; Slocik, J. M.; Naik, R. R. Nano Lett. 2006, 6, 486 984 (6) Daniel, M. C.; Astruc, D. Chem. Rev. 2004, 104, 293. 488 (7) Cao, Y. W.; Jin, R. C.; Mirkin, C. A. J. Am. Chem. Soc. 2001, 123, 489 490 (8) Gill, R.; Polsky, R.; Willner, I. Small 2006, 2, 1037. 491 (9) Chan, S. C.; Barteau, M. A. Langmuir 2005, 21, 5588. 492 (10) Lin, H. Y.; Chen, Y. F.; Wu, J. G.; Wang, D. I.; Chen, C. C. Appl. 493 Phys. Lett. 2006, 88, 161911. (11) Wang, X.; Kong, X.; Yu, Y.; Zhang, H. J. Phys. Chem. C 2007, 495 111, 3836. 496 (12) Ma, G. H.; He, J.; Rajiv, K.; Tang, S. H.; Yang, Y.; Nogami, M. 497 Appl. Phys. Lett. 2004, 84, 4684. 498 (13) Wang, Z. L.; Song, J. Science 2006, 312, 242. 499 (14) Tang, H. X.; Yan, M.; Ma, X. Y.; Zhang, H.; Wang, M.; Yang, D. 500 R. Sens. Actuator B: Chem. 2006, 113, 324. 501 (15) Derfus, A. M.; Chan, W. C. W.; Bhatia, S. N. Nano Lett. 2004, 4, 502 503 (16) Cao, Y. C.; Jin, R.; Mirkin, C. A. Science 2002, 297, 1536. 504 (17) Kneipp, K.; Wang, Y.; Kneipp, H.; Perelman, L. T.; Itzkan, I.; 505 Dasari, R. R.; Feld, M. S. Phys. Rev. Lett. 1997, 78, 1667. 506 (18) El-Sayed, I. H.; Huang, X.; El-Sayed, M. A. Nano Lett. 2005, 5, 507 829. (19) Ozga, K.; Kawaharamura, T.; Umar, A. A.; Oyama, M.; Nouneh, 509 K.; Slezak, A.; Fujita, S.; Piasecki, M.; Reshak, A. H.; Kityk, I. V. 510 Nanotechnology 2008, 19, 185709. 511 (20) Haldar, K. K.; Sen, T.; Patra, A. J. Phys. Chem. C 2008, 112, 512 11650. 513 (21) Chang, S. J.; Hsueh, T. J.; Chen, I. C.; Huang, B. R. 514 Nanotechnology 2008, 19, 175502. 515 (22) Liu, Y.; Zhong, M.; Shan, G.; Li, Y.; Huang, B.; Yang, G. J. Phys. 516 Chem. B 2008, 112, 6484. 517 (23) Shan, G.; Wang, S.; Fei, X.; Liu, Y.; Yang, G. J. Phys. Chem. B 518 2009, 113, 1468. 519 (24) Zhang, W. Q.; Lu, Y.; Zhang, T. K.; Xu, W. P.; Zhang, M.; Yu, S. 520 H. J. Phys. Chem. C 2008, 112, 19872. (25) Chen, B. D.; Zhang, H.; Du, Ning, Li, D. S.; Ma, X. Y.; Yang, D. 522 Mater. Res. Bull. 2009, 44, 889. (26) Wei, Y. Y.; Li, Y.; Liu, X. Q.; Xian, Y. Z.; Shi, G. Y.; Jin, L. T. 524 Biosens. Bioelectron. 2010, 26, 275. (27) Sabramanian, V.; Wolf, E. E.; Kamat, P. V. J. Phys. Chem. B 526 2003, 107, 7479. (28) Wood, A.; Giersig, M.; Mulvaney, P. J. Phys. Chem. B 2001, 105, 528 8810. 529 (29) Vayssieres, L.; Keis, K.; Hagfeldt, A.; Lindquist, S. E. Chem. 530 Mater. 2001, 13, 4395. 531 (30) Yin, S.; Sato, T. J. Mater. Chem. 2005, 15, 4584. 532 (31) Liu, B.; Zeng, H. C. J. Am. Chem. Soc. 2003, 125, 4430. 533 (32) Wang, Z.; Qian, X. F.; Yin, J.; Zhu, Z. K. Langmuir 2004, 20, 534 535 (33) Yang, Y. M.; Lai, H.; Tao, C. Y.; Yang, H. J. Mater. Sci.: Mater. 536 Electron. 2010, 21, 173. 537 (34) Zhang, Y. Y.; Mu, J. J. Colloid Interface Sci. 2007, 309, 478. 538 (35) Parkinson, J. A.; Sun, H. Z.; Sadler, P. J. Chem. Commun. 1998, 539 (36) Tian, Z. R.; Voigh, J. A.; Liu, J.; Mckenzie, B.; Mcdermott, M. J.; 541 Rodriguez, M. A.; Konishi, H.; Xu, H. F. Nat. Mater. 2003, 2, 821. 542 (37) Xu, L.; Guo, Y.; Liao, Q.; Zhang, J.; Xu, D. J. Phys. Chem. B 543 2005, 109, 13519. 544 (38) Liang, J. B.; Liu, J. W.; Xie, Q.; Bai, S.; Yu, W. C.; Qian, Y. T. J. 545 Phys. Chem. B 2005, 109, 9643. 546 (39) Caswell, K. K.; Bender, C. M.; Murphy, C. J. Nano Lett. 2003, 3, 547 548 (40) Liang, J. B.; Bai, S.; Zhang, Y. S.; Li, M.; Yu, W. C.; Qian, Y. T. J. 549 Phys. Chem. C 2007, 111, 1113. 550 (41) Qian, H. F.; Li, X. Q.; Ren, J. C. J. Phys. Chem. B 2006, 110, 551 (42) (a) Yang, P.; Yan, H.; Mao, S.; Russo, R.; Johnson, J.; Saykally, 553 R.; Morris, N.; Pham, He, J. R.; Choi, H. J. Adv. Funct. Mater. 2002, 12, 554

- 555 319. (b) Wu, J. J.; Liu, S. C. Adv. Mater. 2002, 14, 215. (c) Li, Y. B.;
- 556 Bando, Y.; Golberg, D. Appl. Phys. Lett. 2004, 84, 3603.
- 557 (43) (a) Wood, A.; Giersig, M.; Mulvaney, P. J. Phys. Chem. B 2001,
- 558 105, 8810. (b) Subramanian, V.; Wolf, E. E.; Kamat, P. V. J. Phys. 559 Chem. B **2003**, 107, 7479.
- 560 (44) (a) Bai, X. D.; Wang, E. G.; Gao, P. X.; Wang, Z. L. Nano Lett.
- 561 2003, 3, 1147. (b) Wang, X. D.; Summers, C. J.; Wang, Z. L. Appl.
- 562 Phys. Lett. 2005, 86, 013111.
- 563 (45) Thayer, K. A.; Melnick, R.; Burns, K.; Davis, D.; Huff, J. Environ. 564 Health Perspect. **2005**, 113, 1271.
- 565 (46) Su, Y. Y.; He, Y.; Lu, H. T.; Sai, L. M.; Li, Q. N.; Li, W. X.;
- 566 Wang, L. H.; Shen, P. P.; Huang, Q.; Fan, C. H. Biomaterials 2009, 30, 567 19.
- 568 (47) Li, K. G.; Chen, J. T.; Bai, S. S.; Wen, X.; Song, S. Y.; Yu, Q.; Li, 569 J.; Wang, Y. Q. *Toxicol. in Vitro* **2009**, 23, 1007.
- 570 (48) (a) Lovric, J.; Bazzi, H. S.; Cuie, Y.; Fortin, G. R. A.; Winnik, F.
- 571 M.; Maysinger, D. J. Mol. Med. 2005, 83, 377. (b) Sasanka, D.;
- 572 Alessandra, Q.; Grazia, L. M. J. Am. Chem. Soc. 2009, 131, 2948.