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Structures and Crystal Chemistry of Li₂BNH₆ and Li₄BN₃H₁₀

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> Received November 21, 2007 Revised Manuscript Received January 8, 2008

Intensive investigations have been undertaken to search for new light metal hydrides and complex hydrides that possess high hydrogen-storage capacities. Among these, a series of studies in (1-x)LiBH₄ - xLiNH₂ systems have attracted great interest. 1-3 At least four quaternary complex hydride phases with distinct crystalline diffraction peaks have been observed.² Many efforts, both experimentally³⁻⁶ and theoretically^{7,8} have been made to determine the crystal structure of the most stable phase, Li₄BN₃H₁₀, which was observed in a range of x = 0.66-0.75. The previously reported structures determined using X-ray and neutron diffraction on hydride samples possess identical space groups and similar lattice parameters. However, the atomic coordinate differences result in bond length discrepancies as large as 0.2 Å. Such unusual bond lengths and the resulting deformed anion groups, if they were true, would cause extra electron transfer and consequently lead to significant changes in the electronic structures, bondings, coordination environments, structure-stability, and properties of this ionic compound. Another composition, Li₂BNH₆ (x = 0.5), was also reported to be stable as a single or major phase in the XRD pattern. However, the exact crystal structure has not been correctly determined. Therefore, to understand the composition-dependent structure variations and reveal the potentially interesting chemistry, it is critical to obtain accurate structure information, especially for the hydrogen atoms, which requres high-resolution neutron diffraction data on isotopically labeled (i.e., ⁷Li, ¹¹B, and D) samples.

Herein, we present the crystal structures and the neutron vibrational spectra of the quaternary hydrides Li₂BNH₆ and Li₄BN₃H₁₀. Combined high-resolution neutron powder diffraction (NPD), neutron vibrational spectroscopy (NVS) and first-principles calculations allow us to gain more insight into both the structure and dynamics of these novel hydride phases. The NPD samples were isotopically labeled with compositions of ⁷Li₂¹¹BND₆ and ⁷Li₃¹¹BN₂D₈. Such labeling avoids the large neutron absorption from natural Li and B and the high incoherent scattering background from H. The compounds were synthesized by mixing stoichiometric ratios (1:1 and 1:2) of isotopically labeled borohydride and amide precursors and afterward annealing at 338 and 413 K, respectively. See the Supporting Information for details.

The crystal structures of Li₂BNH₆ and Li₄BN₃H₁₀ were solved and refined from NPD data. For both isotopically labeled samples, all independent atomic positions, thermal displacements, and lattice parameters were refined without any constraints. The final Rietveld fit of NPD patterns showed excellent goodness of fit and thus gave us explicit and accurate information on the crystal structure, local atomic coordinations, and bond lengths. Details about the structure solution and refinement are in the Supporting Information. For simplicity, we use general symbols Li, B, and H instead of the isotopes ⁷Li, ¹¹B, and D in the discussion below. We also emphasize that the refined structures on the isotopically labeled samples essentially do not differ from those with natural elements.

In both Li₂BNH₆ and Li₄BN₃H₁₀, the geometry of the BH₄⁻ tetrahedra and NH₂⁻ resulting from the present refinements have nearly ideal configurations. For Li₂BNH₆, the mean B-H bond length is 1.208 Å and N-H bond lengths are 0.990(7) and 1.034(7) Å. For $Li_4BN_3H_{10}$, the mean B-H bond length is 1.195 Å and N-H bond lengths are 0.996(6) and 1.002(4) Å. All these differ markedly from those in the previous XRD and neutron studies, where the N-H bond lengths were found to be anomalously short, 0.83 Å to 0.86 Å, or the NH₂⁻ groups were abnormally deformed with an H-N-H bond angle of 126° and a significant difference of N-H lengths as large as 0.1 Å, and the BH₄ tetrahedra were deformed with one B-H significantly shorter than the other three. The dimensions and configurations of $\mathrm{BH_4}^-$ and $\mathrm{NH_2}^-$ in the present study are found to be close to those in ⁷Li¹¹BD₄ (nearly ideal tetrahedra with a mean B-H distance of 1.214 Å), and to those in $LiND_2^{10}$ (0.99 and 1.00 Å), respectively. These bond lengths in Li₄BN₃H₁₀ also compare favorably with those from our own and others first-principles calculations. Interestingly, our refined B-H bond lengths indicated that the size of BH₄ slightly decreases with its concentration in the structure, i.e., from LiBH₄ to Li₂BNH₆ to Li₄BN₃H₁₀, the mean B-H length changes from 1.214 to 1.208 Å and to 1.195 Å. After our study on the isotopically labeled samples, we have noticed

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a very recent structural study on Li₂BNH₆ from a sample without isotope enrichment. 11 The NH₂ in this structure was found abnormally deformed and the hydrogen positions were declared unreliable due to the poor quality of the neutron data. The apparent discrepancies and the largely deformed BH₄ and NH₂ in these studies are due to the light H and Li atoms, complicating artifacts of X-ray interactions with strong N-H bond polarization, the high neutron absorption cross sections associated with naturally occurring Li and B, and the large incoherent neutron scattering cross section of H. Therefore, the B-H and N-H bond lengths from the present NPD study using isotopically labeled samples are more reliable. In addition, we also note that recent NPD structure refinements on samples without ¹¹B enrichment provide incorrect B-H bond lengths and an anomolously short Li-B distance (1.73 Å). Our determined Li₂BNH₆ and Li₄BN₃H₁₀ structures provided a good structural starting point for the theoretical calculations on the electronic structures, phonon density of states, and reaction energetics.

Li₂BNH₆ contains two symmetry-independent Li⁺ ions, each coordinated with four anion units (Figure 1). Li1 is surrounded by 3 BH₄ and 1 NH₂. Li2 is coordinated by 3 NH₂⁻ and 1 BH₄⁻. Li₄BN₃H₁₀ has three symmetryindependent Li⁺ ions, each also tetrahedrally coordinated. Li1 is surrounded by 2 BH₄⁻ and 2 NH₂⁻, Li2 by 3 NH₂⁻ and 1 BH₄⁻, and Li3 by 4 NH₂⁻. Interestingly, Li⁺ in LiBH₄ and LiNH₂ are all tetrahedrally coordinated by either 4 BH₄⁻ or 4 NH2⁻. The Li-B and Li-N distances in these compounds are summarized in Table S7 in the Supporting Information. In all these compounds, each anion unit is also surrounded by 4 Li⁺. From Figure 1 (top two rows, right), Li₄BN₃H₁₀ can be generated from LiNH₂ through the partial substitution of NH₂⁻ by BH₄⁻ (0, ½, and ½ of NH₂⁻ in one Li[NH₂⁻]₄-tetrahedron). The local coordination environments of Li⁺ in LiNH₂ (there are 8 Li⁺ in a unit cell) were changed with BH₄ substitution, resulting in tetrahedra in one unit cell that are 3/8 Li[NH₂⁻]₄, 2/8 Li[NH₂⁻]₃[BH₄⁻]₁, and 3/8 Li[NH₂⁻]₂[BH₄⁻]₂. Thus, the consequential arrangement with "Li-tetrahedra ordering" can actually be described by a "Li₃₂[BH₄]₈[NH₄]₂₄" cubic supercell, four times the original LiNH₂ tetragonal unit cell ($a_c \approx 2a_t \approx c_t$). The larger lattice parameter of Li₄BN₃H₁₀ is apparently due to the larger size of BH₄⁻. With more BH₄⁻ substitution, the direct replacement of NH₂⁻ by BH₄⁻ would result in a too short distance between neighboring BH₄ ions. Therefore, the lattice has to make a concerted rearrangement to accommodate the larger BH₄ anion unit in the structure and maintain an appropriate BH₄⁻ distance. Figure 2 compares the structure of Li₂BNH₆ and Li₄BN₃H₁₀. The similarities in the structures imply that Li₂BNH₆ can be viewed as an approximate derivative of Li₄BN₃H₁₀ by the transformation $a_h \approx$ $\sqrt{2}a_c$, $c_h \approx \sqrt{3}/2a_c$. Given the BH₄ and NH₂ ratio of 18:18 in the hexagonal unit cell, and the volumes of LiBH₄ (52.4 Å³, 5 K) and LiNH₂ (31.9 Å³, 5 K) formula units, the calculated volume of Li₂BNH₆ should be 1516.5 Å³, 107.8 Å³ smaller than the observed value, leaving large voids in the structure. Compared to the relatively densely packed Li-

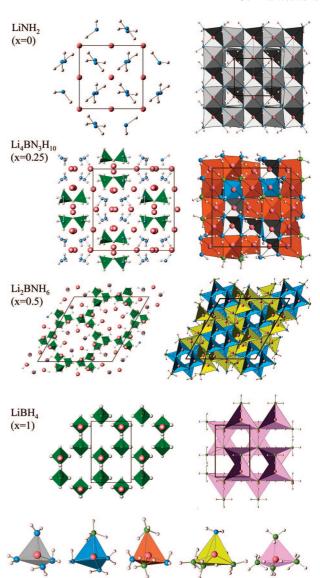


Figure 1. Crystal structure of Li[BH₄]_x[NH₂]_{1-x}. In each row, the structure is shown in both a ball-stick view (left) containing BH₄⁻ tetrahedral units (in green) and NH₂⁻ units; and as a polyhedral diagram (right) with Lianion tetrahedra colored according to the different coordination environments of Li. (Bottom row) different Li-anion tetrahedra configurations existing in the structures of these complex hydrides. The Li[NH₂⁻]₄ tetrahedron is shown in grey, Li[NH₂⁻]₃[BH₄⁻]₁ in blue, Li[NH₂⁻]₂[BH₄⁻]₂ in orange, Li[NH₂⁻]₁[BH₄⁻]₃ in yellow, and Li[BH₄⁻]₄ in pink. Li⁺ ions are shown as red spheres.

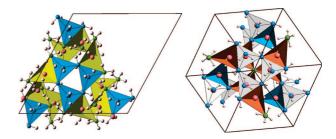


Figure 2. Comparison of structures of (left) Li₂BNH₆ (only half-shown for clarity) and (right) Li₄BN₃H₁₀ (viewed in the [111] direction), showing the structural progression upon the replacement of NH₂⁻ by BH₄⁻ and the resulting changes in the Li[NH₂⁻]_{4-n}[BH₄⁻]_n tetrahedra, n = 0, 1, 2, 3. The Li tetrahedra with differently combined anions were colored as in Figure 1

anion tetrahedra arrays in Li₄BN₃H₁₀, such large voids in Li₂BNH₆ suggested a less stable structure. Annealing

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Li₂BNH₆ at 363 K resulted in decomposition into Li₄BN₃H₁₀ and LiBH₄. In addition, a previous phase diagram study indicated that Li₂BNH₆ has a more rigid stoichiometry than the Li₄BN₃H₁₀ phase, which was found to exist over a wide solid solution range.

In seeking an explanation for the rigid stoichiometry of Li₂BNH₆, it is useful to consider the local bonding configurations in all possible types of {Li[BH₄]_n- $[NH_2^-]_{4-n}$ tetrahedra, n = 0-4. Among them, Li $[BH_4^-]_2$ -[NH2⁻]₂ shows the largest distortion due to the largest difference between BH₄ and NH₂ components (i.e., the largest difference between Li-B and Li-N distances). Such incompatability can be accommodated in the Li₄BN₃H₁₀ structure through a small displacement of the neighboring ions. For Li₂BNH₆, as mentioned above, Li[BH₄⁻]₁[NH₂⁻]₃ and Li[BH₄⁻]₃[NH₂⁻]₁ tetrahedra are present. Replacement of NH₂⁻ by BH₄⁻ in the former configuration would yield Li[BH₄⁻]₂[NH₂⁻]₂, which would severely compromise the local stability and could not be significantly improved by the displacement of its neighbors because its neighboring tetrahedra have also been converted into this same configuration. Replacement of $\mathrm{NH_2}^$ by BH₄ Li[BH₄⁻]₃[NH₂⁻]₁ would not only induce the formation of $\text{Li}[BH_4^-]_2[NH_2^-]_2$ but also $\text{Li}[BH_4^-]_4$. In the latter case, the Li⁺ has to move toward the original three BH₄⁻ to accommodate the new large anion unit, thus further reducing the overall energetic stability. As a result, Li[BH₄⁻]₂[NH₂⁻]₂ tetrahedra are completely avoided in the actual structure of Li₂BNH₆. In general, any BH₄⁻-rich quaternary compositions in the LiBH₄-LiNH₂ system are expected to be unstable and indeed have not been observed. Complete substitution of NH₂ by BH₄ results in the formation of LiBH₄ with the presence of Li[BH₄]₄ tetrahedra exclusively.

Considering the structure variation from LiNH2 to Li₂BNH₆, these novel quarternary Li-B-N-H phases can actually be viewed as solid solutions with the general formula $Li[BH_4]_x[NH_2]_{1-x}$ via a gradual substitution of BH_4 for NH₂⁻ in the binary end member LiNH₂ (see Figure 1). Such a trend is also evident by the increased size of BH₄⁻ ions with increasing x. The structures of all these phases can thus be generalized as three-dimensional frameworks consisting of corner- and face-shared metal-anion tetrahedra, $\{Li[BH_4^-]_n[NH_2^-]_{4-n}\}, n = 0-4, \text{ with Li in the center and } 1$ various combinations of BH₄ and NH₂ units at the vertices. A prior phase diagram study showed there are at least four quarternary phases in LiBH₄-LiNH₂. However, the structure investigation of these phases are complicated by their instabilities and uncertainties of stoichiometry. From the present study, we believe the general view of the 3D structural construction with $\{Li[BH_4^-]_n[NH_2^-]_{4-n}\}$ tetrahedra as building units actually provides us new clues to determine the structure and the stoichiometry of the remaining unknown phases.

The neutron vibrational spectra of Li₄BN₃H₁₀ and Li₂BNH₆ are shown in Figure 3. The similarities in the spectra indicate some similarities in their structure. According to our first-

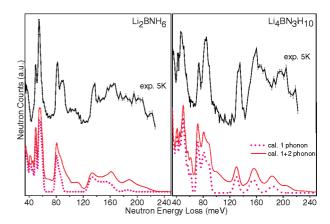


Figure 3. Neutron vibrational spectra and calculated phonon modes of Li₂BNH₆ and Li₄BN₃H₁₀.

principles phonon calculations, the observed bands can be assigned to N-H (80-90 meV, 180-190 meV) and B-H (130–165 meV) bending and deformation modes, NH₂⁻ and BH₄ collective and independent librational modes (53–65 meV), and low-frequency lattice modes. Some vibrations of BH₄⁻ and NH₂⁻ in the energy range below 60 meV are also found to be coupled with the motions of Li⁺. The calculated phonon modes agree reasonably well with the observed NV spectra for both compounds and thus are consistent with our determined structures. From the calculations, the B-H stretching modes in these quaternary hydrides are within the same B-H stretching region of LiBH₄ and Mg(BH₄)₂, ¹² suggesting similar B-H bond lengths in these borohydrides, in agreement with the refined B-H distances.

In summary, the crystal structures of Li₂BNH₆ and Li₄BN₃H₁₀ were correctly determined and refined using highresolution neutron powder diffraction data on ⁷Li₂¹¹BND₆ and ⁷Li₄¹¹BN₃D₁₀. The present structures explicitly show the nearly ideal configurations for both BH₄⁻ and NH₂⁻, clarifying the large discrepancies in the prior reports on artificial anion distortions and inaccurate bond distances. These novel quarternary Li-B-N-H phases can be viewed as 3D frameworks with a general solid-solution formula Li[BH₄]_x[NH₂]_{1-x}, built up by corner- and face-shared $\{Li[BH_4^-]_n[NH_2^-]_{4-n}\}$ tetrahedra and formed via a gradual substitution of BH₄⁻ for NH₂⁻ from LiNH₂ to LiBH₄.

Acknowledgment. We thank Prof. Michael Hartman (University of Michigan) for generously providing ⁷Li¹¹BD₄. This work was partially supported by the DOE through EERE Grant DE-AI-01-05EE11104 (T.J.U.) and BES Grant DE-FG02-98ER45701 (T.Y.)

Supporting Information Available: Details on materials synthesis, neutron experiments, structure data, and first-principles calculations; summary of all the reported Li₄BN₃H₁₀ structures (PDF). This material is available free of charge via the Internet at http://pubs.acs.org.

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