

QUANTUM ESPRESSO (QE)



- > Suite of open-source computer codes for first-principles (DFT) electronic-structure calculations
- > ESPRESSO = opEn-Source Package for Research in Electronic Structure, Simulation and Optimization
- > Plane wave (PW) basis set & pseudopotentials

QUANTUM ESPRESSO (QE)



Languages



Roff 7.2% • C 5.0%

- PostScript 4.2%
 HTML 3.7%
- Shell 3.2%
 Other 6.8%

Over 300,000 lines of code

Journal of physics: Condensed matter, 2009, 21.39: 395502. Journal of physics: Condensed matter, 2017, 29.46: 465901

WHAT CAN QE DO?



- 1. Ground-state calculations (energies, forces, stresses, Kohn-Sham orbitals, ...)
- 2. Structural optimizations
- Response properties (phonons, infrared & Raman cross-sections, ...)
- Spectroscopic properties (X-ray, electronic excitations, ...)
- 5. Quantum transport
- 6. ...

CORE PACKAGES



- PWscf = Plane-Wave Self-Consistent Field (pw.x)
- \triangleright CP = Car-Parrinello Molecular Dynamics (cp.x)

SPECIALIZED PACKAGES



- PostProc = Data postprocessing (pp.x)
- \triangleright PHonon = Phonons with Density-Functional Perturbation Theory (ph.x)
- > ...

USEFUL RESOURCES



Documentation:

https://www.quantum-espresso.org/documentation/

Mailing list:

https://www.mail-archive.com/users@lists.quantum-espresso.org/

> Tutorials:

https://www.quantum-espresso.org/tutorials/

➤ QE paper on arXiv:

https://arxiv.org/abs/0906.2569

Journal of physics: Condensed matter, 2009, 21.39: 395502. Journal of physics: Condensed matter, 2017, 29.46: 465901

https://www.quantum-espresso.org/

XCRYSDEN



- XcrySDen is crystalline and molecular structure vizualization program
- Compatible with QE input/output files
- Can be used to define k-point paths and grids in the Brillouin zone

WHERE TO FIND (CRYSTAL) STRUCTURES?

Cambridge Crystallographic Data Centre:

https://www.ccdc.cam.ac.uk/

Crystallography Open Database

http://www.crystallography.net/cod/

> DFT calculations database on Materials Project:

https://materialsproject.org/

- Papers
- > If no structures are available, you have to use structure prediction algorithms

- > Input data is organized as several namelists, starting with &
 - > &CONTROL = general variables controlling the run (calculation type, file 10 locations...)
 - > &SYSTEM = structural information on the system under investigation
 - > &ELECTRONS = electronic variables: self-consistency, smearing
 - > &IONS (optional) = ionic variables: relaxation, dynamics
 - \geq &CELL (optional) = variable cell relaxation, dynamics
- Namelists are followed by other fields (cards) introduced by keywords:
 - > ATOMIC_SPECIES, ATOMIC_POSITIONS, K_POINTS, (CELL_PARAMETERS, OCCUPATIONS)

- \triangleright How to choose various parameters (ecutwfc, ecutrho, k-point grid density, ...)?
- They have to large enough so that the results (energies, forces, ...) are converged (don't change a lot upon further increasing of the parameters)
- > Beware of computational cost vs. accuracy

> The next exercise will be on explicitly converging the results

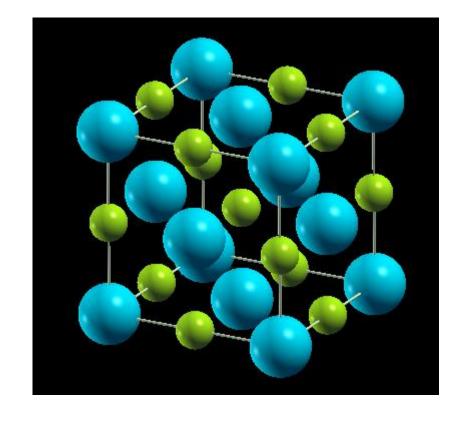
- > The default values of the nbnd variable are
- 1. Insulators = number of valence bands (#electrons / 2)
- 2. Metals = number of valence bands + 20% more (minimum 4 more)
- > The number of electrons is written in the pseudopotential files (Z valence)
- > To view the location of the pseudopotential files, type

echo \$ESPRESSO_PSEUDO

```
ibrav = 2, ! FCC
  celldm(1) = 10.571, ! lattice constant (Bohr)

...

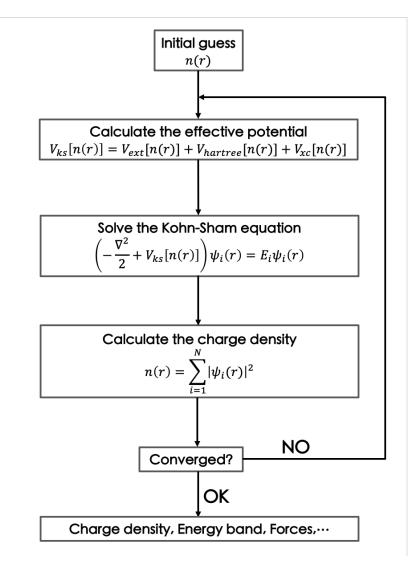
ATOMIC_POSITIONS {crystal}
  Na 0.00 0.00 0.00
  Cl 0.50 0.50 0.50
```



- > After performing the pw.x calculation, we can observe the generated output files:
 - \rightarrow nacl.xml = information about the calculation (for multistep calculations intermediate configurations are saved)
 - > nacl.save = directory containing wave functions, charge density and pseudopotentials
 - > nacl.out = main human-readable output file

- > nacl.out contents by line number:
- \geq 13 35 = Parallelization information
- \geq 39 683 = Information on the system, SCF parameters and symmetries
- \geq 685 708 = k-points list
- > 710 754 = Memory usage
- > 756 1032 = SCF cycles
- \geq 1036 1124 = Forces & stress contributions
- > 1129 1197 = Timing

```
Initial potential from superposition of free atoms
starting charge
                   15.9937, renormalised to
                                               16.0000
negative rho (up, down): 1.484E-02 0.000E+00
Starting wfcs are
                   9 randomized atomic wfcs + 1 random wfcs
total cpu time spent up to now is 1.4 secs
Self-consistent Calculation
iteration # 1
                 ecut=
                         50.00 Ry
                                      beta= 0.70
Davidson diagonalization with overlap
ethr = 1.00E-02, avg # of iterations = 4.7
Davidson diagonalization with overlap
ethr = 3.33E-04, avg # of iterations = 2.3
negative rho (up, down): 1.531E-02 0.000E+00
total cpu time spent up to now is 3.3 secs
total energy = -128.88647981 \text{ Ry}
estimated scf accuracy <
                               0.05091372 Ry
```

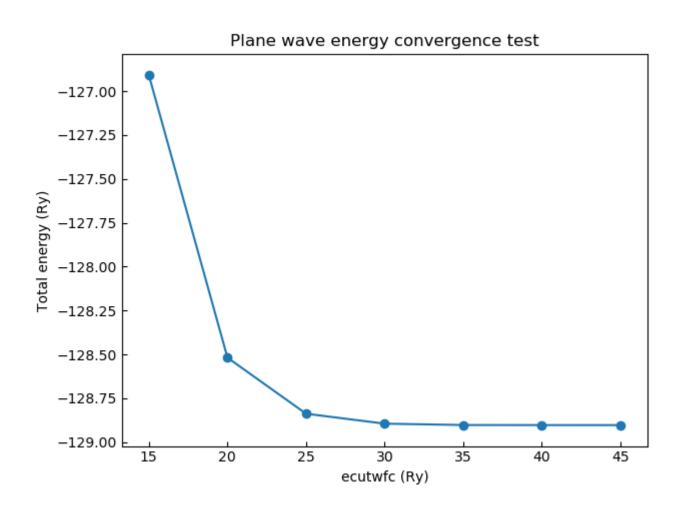


```
total energy
                    = -128.88647981 Ry
                           0.05091372 Ry
total energy
           = -128.90106640 Ry
                           0.01055485 Ry
total energy = -128.90302693 \text{ Ry}
                           0.00022615 Ry
          = -128.90306844 Ry
total energy
                           0.00002059 RV
total energy
            = -128.90307086 Ry
                           0.00000224 Ry
total energy = -128.90307206 \text{ Ry}
                           0.00000065 Ry
total energy
            = -128.90307218 Ry
                             4.2E-09 Ry
total energy
            = -128.90307218 Ry
                             6.8E-10 Ry
total energy
                    = -128.90307218 Ry
estimated scf accuracy
                             6.6E-11 Ry
```

SCF is converged when accuracy < conv_thr</p>

- > We must ensure that the results of a calculation do not depend on free parameters
- > In this exercise, we will look at the convergence of energy with respect to:
 - 1. ecutwfc
 - 2. k-point grid density

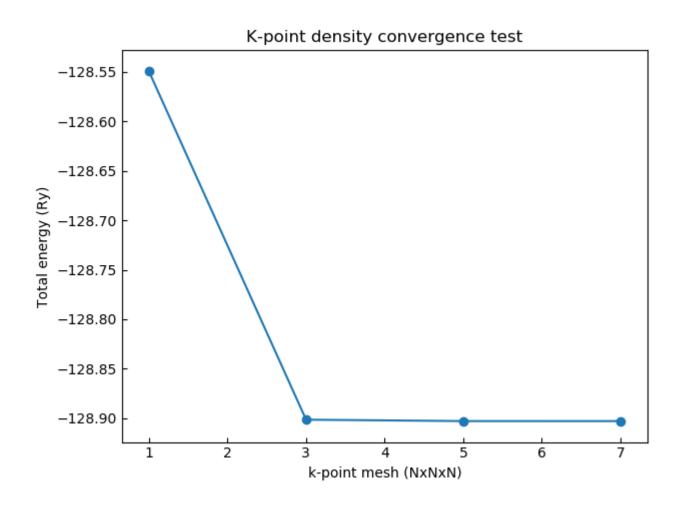
- \triangleright ecutwfc = plane wave (basis set) cutoff energy
- Convergence w.r.t. to ecutwfc is a property of the pseudopotential(s) used
- > Absolute energy typically converges slower than other properties (e.g. structure)



- > k-points can be specified as a list of (specialized) points or a uniform grid
- If a list is chosen (usually for band structure plots): provide a list of k-points in the irreducible BZ and corresponding symmetry weights
- > FAQ: where to find special k-points and their weights?
- Auxillary code (kpoints.x, <u>SeeK-Path</u>)
- 2. Papers
- We will also see how to explicitly select a k-point path in XcrySDen in an another exercise

- > If a uniform grid is chosen, the k-points are generated using the Monkhorst-Pack scheme
- See <u>pw.x documentation</u> for more info on the K_POINTS card
- > Reasonably dense k-point grids can be automatically generated from structure files via

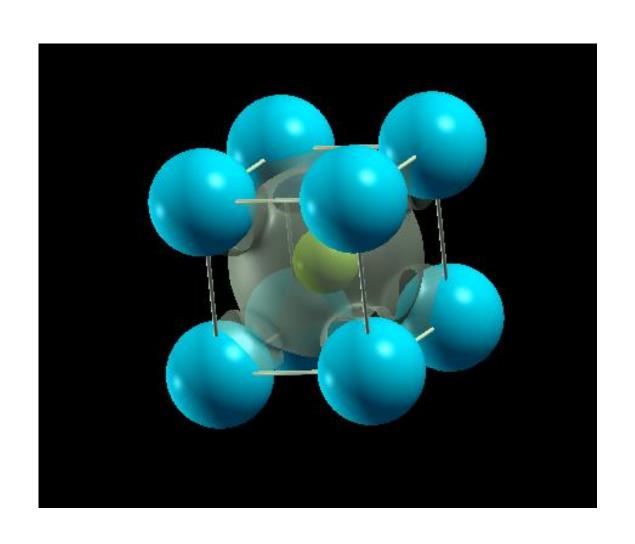
https://www.materialscloud.org/work/tools/qeinputgenerator



- In this exercise, we will introduce the postprocessing program **pp.x**
- > pp.x is used to extract information out of pw.x calculations
- > pp.x input files are structured similarly to pw.x (pp.x documentation)

> To show pp.x basic functionality, we will plot the ground state charge density

- pp.x has two steps
- 1. &INPUTPP = extract the desired data from pw.x output and write to the "filplot" file
- 2. &PLOT = write the extracted data to a desired format
- > Programs for visualization of pp.x output:
- 1. XcrySDen
- 2. <u>Vesta</u> (gaussian .cube files)
- 3. Gnuplot
- 4. ...



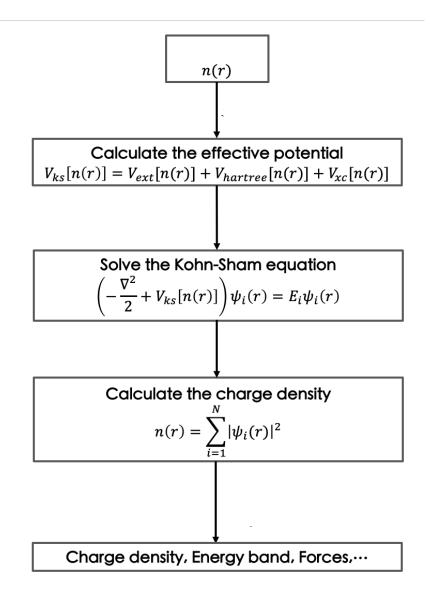
- > In this exercise, we will be using a metallic system (silver) as an example
- > In SCF calculations, it is necessary to perform integration over occupied states
- > The step-like Fermi-Dirac distribution in metals can be a source of numerical issues
- > This is avoided by replacing the FD distribution with a smoother function

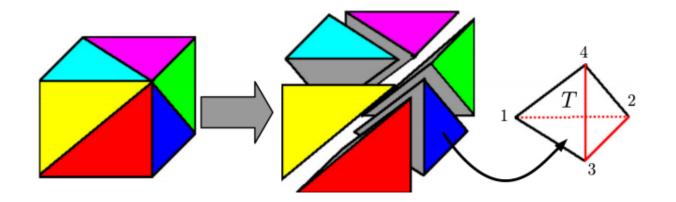
- > Note the presence of three new variables in the pw.x input file:
- 1. occupations = tell QE how to treat the occupied states (fixed, smeared, ...)
- 2. smearing = choose the type of smearing (Methfessel-Paxton)
- 3. degauss = value of the gaussian spreading (Ry) for BZ integration

Results should also be converged w.r.t. the degauss value

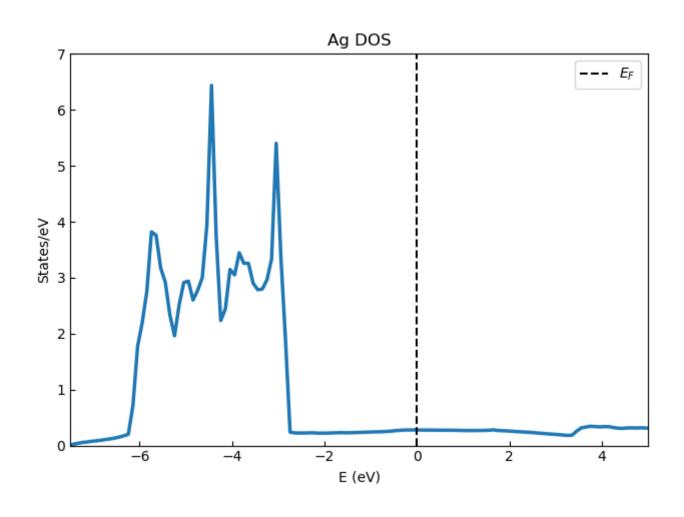
- \triangleright After running pw.x, notice that the Fermi energy is now calculated
- \triangleright The next step is running a non-SCF pw.x calculation with a denser k-point mesh
- \triangleright calculation = 'nscf', occupations = 'tetrahedra'

calculation = 'nscf', occupations = 'tetrahedra'





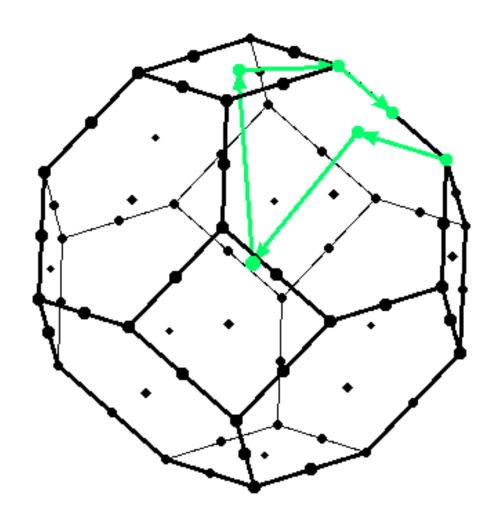
- > The density of states is calculated from the NSCF output using the program dos.x
- Look at the <u>dos.x documentation</u>



- > The usual SCF calculation is followed by an NSCF calculation along a specified path in the 1BZ
- \triangleright calculation = 'bands'
- Useful links for finding special points in the 1BZ:
- SeeK-path
- 2. <u>Bilbao Crystallographic Server</u>
- 3. Wikipedia BZ article

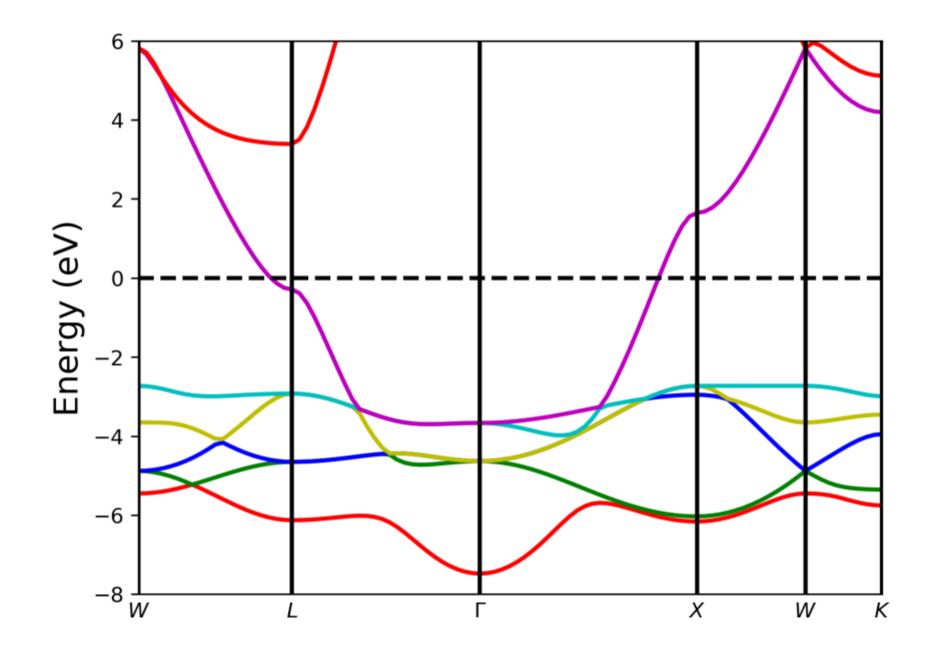
- In this exercise, we will explicitly set the k-path with XcrySDen
- > There is no unique choice of the path, it is often chosen to best represent the bands of interest
- We'll use the path:

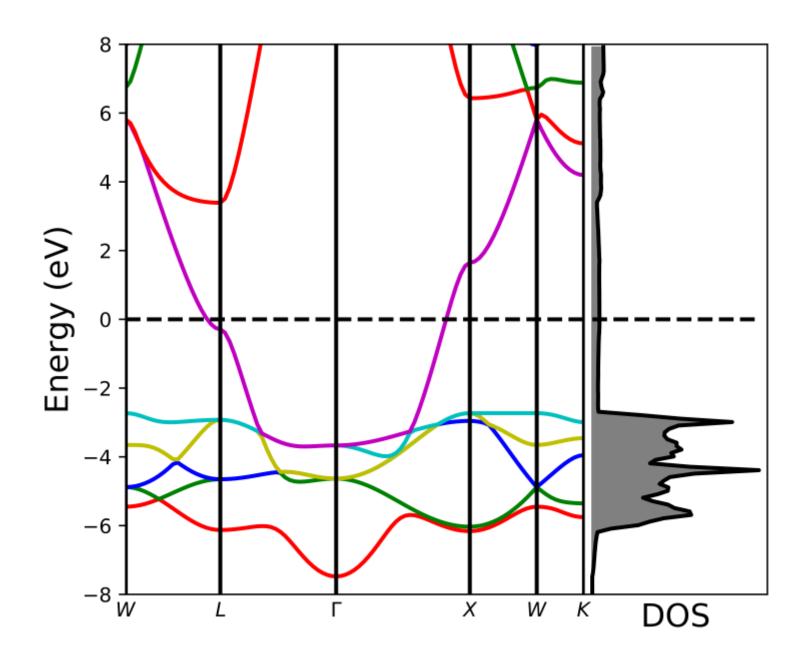
$$W-L-\Gamma-X-W-K$$



Rotation Step: 5		5			
# of Selected Points: 6					
#	reciprocal coordin	nates	label		
1	-0.25000 0.25000	-0.50000	W		
2	0.00000 0.50000	0.00000	L		
3	0.00000 0.00000	0.00000	GAMMA		
4	0.50000 0.50000	0.00000	X		
5	0.25000 0.50000	-0.25000	W		
6	0.00000 0.37500	-0.37500	K		
7					
8					
9					
10					
11					
12					
13					
14					
15					
16					

- The band structure is extracted from the NSCF calculation using the postprocessing program bands.x
- <u>bands.x documentation</u>



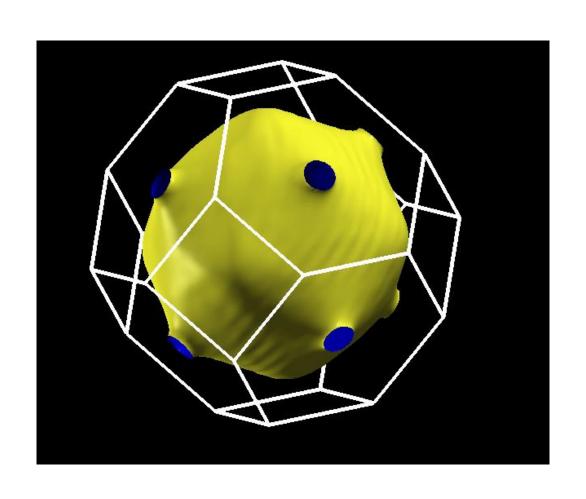


EXERCISE 6 — FERMI SURFACE

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- > Plotting the Fermi surface is a very simple two-step calculation
- SCF calculation
- 2. Extracting the Fermi surface via the postprocessing program fs.x

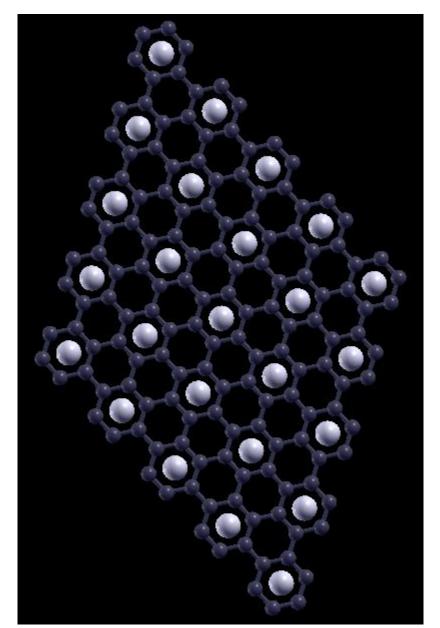
EXERCISE 6 — FERMI SURFACE

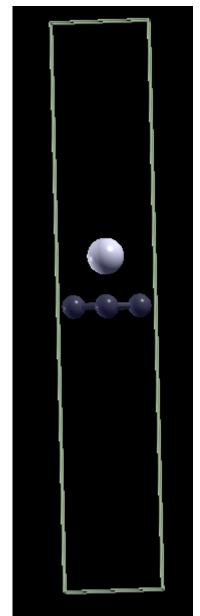


- > In this exercise, we will:
- 1. Calculate the formation energy of a LiC6 two-dimensional layer
- 2. Observe the charge transfer due to the formation of Li-C bonds

- Note the long cell axis in the z-direction
- > This is a way to treat 2D systems

In the following exercise, we will show an alternative way to deal with non-periodic systems





> The formation energy is

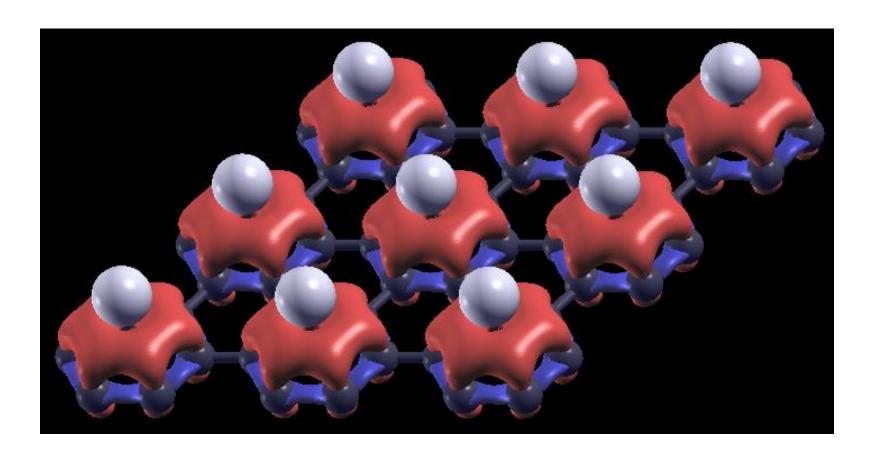
$$E_{\text{form}} = E(\text{LiC}_6) - (E(\text{Li}) + E(\text{C}_6)) \approx -0.0207 \text{ Ry}$$

- \triangleright This means that at T=0 K, adsorption of lithium on a graphene is a favourable reaction
- > Note that this type of calculation doesn't give us any information on the dynamics of the process
- Molecular dynamics (MD), nudge elastic band (NEB), ...

> Charge difference is similarly calculated as:

$$\rho_{\text{diff}} = \rho(\text{LiC}_6) - (\rho(\text{Li}) + \rho(\text{C}_6))$$

- 1. We extract the ground state density of the three systems using pp.x and $plot_num = 0$
- 2. Subtract the charges by specifying nfile = 3 and appropriate weights



 \triangleright Surfaces of constant charge density (red = +0.0025, blue = -0.0025)

EXERCISE 8 — KOHN-SHAM ORBITALS

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- > In this exercise, we will plot the Kohn-Sham orbitals of a single benzene (C6H6) molecule
- QE assumes periodic boundary conditions
- > To simulate isolated (nonperiodic) systems we have two options
- Make a large cell
- 2. Use one of the corrections defined under the assume_isolated variable in the pw.x documentation

EXERCISE 8 — KOHN-SHAM ORBITALS

- \rightarrow The orbitals are extracted from the SCF calculation with pp.x and plot_num=7
- > LUMO is specified as kband(2)=16 because the HOMO orbital is

$$> n_{HOMO} = \frac{N_C Z_C + N_H Z_H}{2} = \frac{6*4 + 6*1}{2} = 15$$

 \succ The number of valence electrons Z_X of C and H atoms is specified in the pseudopotential files

