

ABINIT Hands-On

for Advanced Experimental Physics

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What is ABINIT

- Software to compute properties of materials
- Density-Functional-Theory (DFT) + perturbation theories + ...
- Website: <https://www.abinit.org/>

How to install ABINIT

- **Quantum Mobile:** <https://quantum-mobile.readthedocs.io/en/latest/index.html>
 - pre-configured virtual machine (please, use the last version!) for VirtualBox
 - highly recommended and only option supported by the teachers
- **Conda:** <https://anaconda.org/>
 - package manager for macOS, Windows, Linux
 - ABINIT support for Linux only (and probably for Windows through Windows Linux Subsystem - WLS)
 - Use Conda-Forge version:
 - Add Conda-Forge channel to Conda: `conda config --add channels conda-forge`
 - Create and activate new Conda environment: `conda create -n cf.abinit && conda activate cf.abinit`
 - Install ABINIT: `conda install -c conda-forge abinit`
 - Remember to activate environment (`conda activate cf.abinit`) when you open a new terminal
- **APT:** `sudo apt install abinit`
 - built-in package manager of Debian-based Linux distributions
 - available on Windows through Windows Linux Subsystem (WLS)
- **Compilation:** https://docs.abinit.org/tutorial/abinit_build/
 - recommended only for experienced users
 - NOT supported by the teachers

How to run ABINIT

1. Write input text file **something.abi** where values assigned to (needed) input variables (<https://docs.abinit.org/variables/>)
 2. From terminal, execute **abinit something.abi 1> something.log 2> something.err**
 - **abinit**: command to be executed
 - **something.abi**: path to the input file
 - **1> something.log**: save terminal standard output to **something.log** file (debugging)
 - **2> something.err**: save terminal error output to **something.err** file (debugging)
 3. Several output files generated:
 - **something.abo**: main output file summarizing the data
 - ***_GSR.nc, *_EIG.nc, ...**: output data files
see <https://docs.abinit.org/guide/abinit/#32-file-names-in-abinit> for more info
- To speed up computation, you can run ABINIT with parallel execution. For example, if you want to use 4 cores: **mpirun -n 4 abinit something.abi 1> something.log 2> something.err**

Useful tools

- Certified pseudo-potentials database: <http://www.pseudo-dojo.org/>
- Grace: <http://plasma-gate.weizmann.ac.il/Grace/>
 - band structure and density-of-states plots with the commands `xmgrace *.agr` and `xmgrace *DOS`
 - already included in Quantum Mobile, can be installed with `sudo apt install grace`
- XCrysDen: <http://www.xcrysden.org/>
 - crystal structure visualization
 - Fermi surface plots with the command `xcrysden --bxsf *BSXF`
 - already included in Quantum Mobile
- AFLOW: <http://www.aflowlib.org/>
 - database of material properties (you can download crystal structures from here)
- AbiPy: <https://abinit.github.io/abipy/>
 - Python library with a lot of post-processing options for ABINIT output data
 - not included in Quantum Mobile, but can be installed in Quantum Mobile with Conda
 - very extensive, takes some time to learn and it is not necessary for the course

Useful resources (1/2)

- These slides are based on the following ABINIT tutorials, containing much more information, which you can refer to if you are struggling:
 - ground-state calculations, structural relaxation, electronic properties, SOC:
 - <https://docs.abinit.org/tutorial/base1/>
 - <https://docs.abinit.org/tutorial/base2/>
 - <https://docs.abinit.org/tutorial/base3/>
 - <https://docs.abinit.org/tutorial/base4/>
 - <https://docs.abinit.org/tutorial/spin/>
 - phonon properties:
 - <https://docs.abinit.org/tutorial/rf1/>
 - <https://docs.abinit.org/tutorial/rf2/>
 - electron-phonon coupling:
 - https://docs.abinit.org/tutorial/eph_legacy/

Useful resources (2/2)

- Introductory guide to ABINIT:
 - <https://docs.abinit.org/>
- Slightly more advanced guide to ABINIT, including explanation of input and output files, and multi-dataset mode (not straightforward, hence reading is recommended):
 - <https://docs.abinit.org/guide/abinit/#34-the-multi-dataset-mode>
- Explanation of ABINIT and ANADDDB variables:
 - <https://docs.abinit.org/variables/>
- ABINIT course (very advanced):
 - <https://school2019.abinit.org/course-material>

What are we going to do with ABINIT

- Calculation of the properties of crystalline solids:
 - ground-state calculation
 - structural relaxation
 - electronic properties: band structure, density-of-states (DOS), Fermi surface
 - phonon properties: linear response, inter-atomic force constants, dispersion relation, DOS
 - electron-phonon coupling: matrix elements, superconducting properties

Ground-state calculation

Ground-state (GS) energy

- For a given crystal structure (atomic arrangement):

1. Solve Kohn-Sham eigenvalue problem: $\hat{H}_{\text{KS}} \phi_{n\mathbf{k}}(\mathbf{r}) = \varepsilon_{n\mathbf{k}} \phi_{n\mathbf{k}}(\mathbf{r})$

- $\hat{H}_{\text{KS}} = \left[-\frac{\hbar^2 \nabla^2}{2m_e} + U_{\text{ext}}(\mathbf{r}) + U_{\text{H}}(\mathbf{r}) + U_{\text{xc}}(\mathbf{r}) \right]$
 - $U_{\text{ext}}(\mathbf{r})$: potential energy due to electron-nuclei interaction
 - $U_{\text{H}}(\mathbf{r})$: potential energy due to electron-electron interaction (Hartree potential)
 - $U_{\text{xc}}(\mathbf{r})$: exchange-correlation potential energy
 - $n\mathbf{k}$: band index and electron wave vector (Bloch theorem)

2. Ground-state energy: $E_0 = \sum_{n\mathbf{k}}^{\text{occ}} \varepsilon_{n\mathbf{k}} - \frac{e^2}{2} \int d\mathbf{r} d\mathbf{r}' \frac{n(\mathbf{r})n(\mathbf{r}')}{|\mathbf{r}-\mathbf{r}'|} + E_{\text{xc}}[n] - \int d\mathbf{r} n(\mathbf{r}) U_{\text{xc}}(\mathbf{r})$

- $n(\mathbf{r}) = \sum_{n\mathbf{k}}^{\text{occ}} |\phi_n(\mathbf{r})|^2$: electron density
- $E_{\text{xc}}[n]$: exchange-correlation functional ($U_{\text{xc}} = \delta E_{\text{xc}} / \delta n$)

Electron-nuclei potential energy

- Kohn-Sham eigenvalues problem:

$$\left[-\frac{\hbar^2 \nabla^2}{2m_e} + U_{\text{ext}}(\mathbf{r}) + U_{\text{H}}(\mathbf{r}) + U_{\text{xc}}(\mathbf{r}) \right] \phi_{n\mathbf{k}}(\mathbf{r}) = \varepsilon_{n\mathbf{k}} \phi_{n\mathbf{k}}(\mathbf{r})$$

- $U_{\text{ext}}(\mathbf{r}) = \sum_{\mathbf{R}\alpha} U_{\alpha}(\mathbf{r} - \mathbf{r}_{\alpha} - \mathbf{R})$
 - $\mathbf{R} = n_1 \mathbf{R}_1 + n_2 \mathbf{R}_2 + n_3 \mathbf{R}_3$: Bravais lattice translation vector
 - α : label/index of the atom within unit cell
 - $\mathbf{r}_{\alpha} = x_{\alpha,1} \mathbf{R}_1 + x_{\alpha,2} \mathbf{R}_2 + x_{\alpha,3} \mathbf{R}_3$: atom position within unit cell (fractional coordinates)
 - U_{α} : atomic (pseudo-)potential

Electron-nuclei potential energy in ABINIT

- Inside ABINIT input file **something.abi**:

```
#####
# PREAMBLE
#####
outdata_prefix "./scf/scf" # path prefix for output data files
#####
# UNIT CELL
#####
acell 10.18 10.18 10.18
rprim 0.0 0.5 0.5
      0.5 0.0 0.5
      0.5 0.5 0.0
      } R_i/L_i } {R_1,R_2,R_3}
#####
# ATOM TYPES
#####
ntypat 1
znuc1 14
pp_dirpath "../../../Pseudos/PBE-SR"
pseudos "Si.psp8"
#####
# ATOMS
#####
natom 2
typat 1 1
xred 0 0 0
     1/4 1/4 1/4
```

Annotations in the image:

- Blue arrows point from L_i to the `acell` line and from R_i/L_i to the `rprim` line.
- Blue arrows point from Z_i to the `znuc1` line and from U_i to the `pseudos` line.
- Blue arrows point from i_α to the `typat` line and from x_α to the `xred` line.
- Blue arrows point from $A = \text{natom}$ to the `natom` line and from i_α to the `typat` line.

Here you tell ABINIT that there are 2 atoms in the unit cell.

The first atom is of type 1 ($i_1 = 1$), the second atom is of type 1 ($i_2 = 1$).

Therefore, the first **znuc1** Z_1 and **pseudos** U_1 are associated to both atoms.

Bravais lattice specification

- Using **acell** and **rprim** is not the only way to specify the Bravais lattice underlying your crystal
- An alternative workflow that you might find easier is:
 - Find your material at <http://www.aflowlib.org/search/>
(!) Be careful to choose the correct structure (space group)!

Si [08ab41c5f54850db]

$Fd\bar{3}m$ (#227)

cF8

[API, Out, JSON]

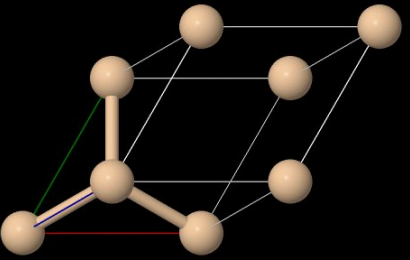
- Select “Standard primitive” unit cell
- Specify unit vector lengths and angles in degrees with the variables **acell** and **angdeg**

```
acell 7.31 7.31 7.31  
angdeg 60 60 60
```

(!) Remember, **acell** should be specify in Bohr units

AFLOW.org consortium (AFLOW v3.2.9)
entry=Si1_ICSD_652255

Spacegroup = $Fd\bar{3}m$ (#227)
 $a=3.87\text{\AA}$, $b=3.87\text{\AA}$, $c=3.87\text{\AA}$
 $\alpha=60^\circ$, $\beta=60^\circ$, $\gamma=60^\circ$



Structure

Relaxed Structure:

[info]

[info]

Visualization:

Axis:

☐ Rotation ☐ Labels

Save Structure:

Plane-wave expansion (in ABINIT)

- Bloch theorem: $\phi_{n\mathbf{k}}(\mathbf{r}) = \sum_{\mathbf{G}} c_{n\mathbf{k}}(\mathbf{G}) \exp i(\mathbf{k} + \mathbf{G}) \cdot \mathbf{r}$
 - \mathbf{G} : reciprocal lattice translation vectors
- Plane-wave representation of Kohn-Sham eigenvalue equation to describe problem in matrix form (numerical solution by calculator):


$$\hat{H}_{\text{KS}} \phi_{n\mathbf{k}} = \varepsilon_{n\mathbf{k}} \phi_{n\mathbf{k}}(\mathbf{r}) \leftrightarrow \mathbf{H}_{\text{KS}}(\mathbf{k}) \mathbf{c}_{n\mathbf{k}} = \varepsilon_{n\mathbf{k}} \mathbf{c}_{n\mathbf{k}}$$

- Finite number of plane waves ($\Rightarrow n = 1, \dots, n_{\text{max}}$) defined by cut-off energy E_{cut} :

$$T = \frac{\hbar^2}{2m_e} |\mathbf{k} + \mathbf{G}|^2 < E_{\text{cut}}$$

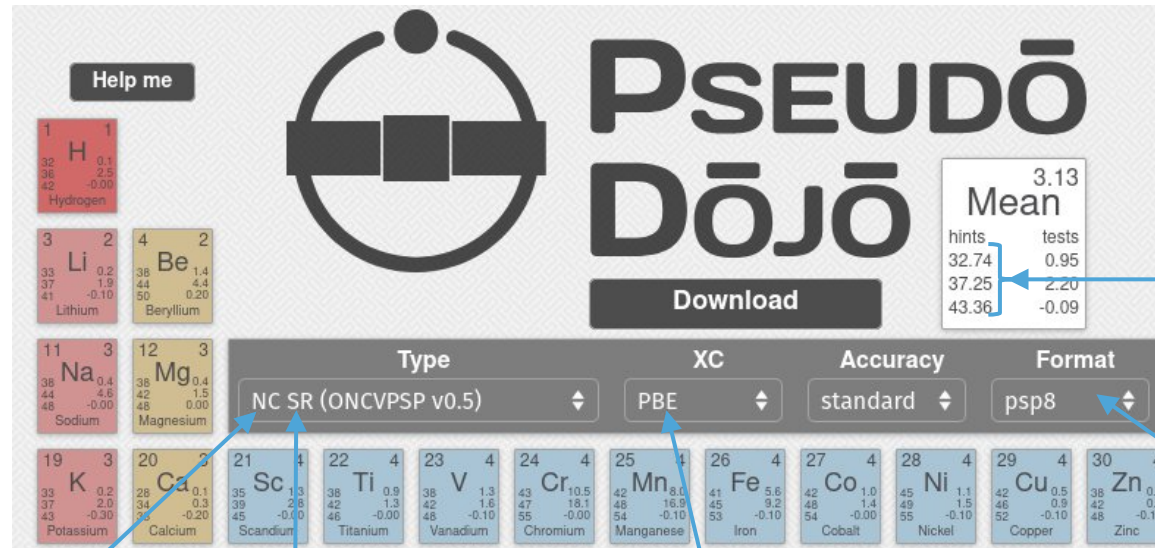
- Inside ABINIT input file **something.abi**:

```
#####  
# PLANE-WAVE BASIS SET  
#####  
ecut 12.0 # [Ha] plane-wave cut-off energy  
# (recommended values at http://www.pseudo-dojo.org/)
```



Pseudo-potentials and plane-wave cut-off

- ABINIT “certified” repository of pseudo-potentials: <http://www.pseudo-dojo.org/>
- Careful to download the appropriate pseudo-potential!



Recommended plane-wave cut-off energy in Hartree, for:

- quick calculation/starting point
- good guess for high-throughput
- high-accuracy

Norm-Conserving
(do not use PAW!)

Scalar-Relativistic
(Full-Relativistic – FR – for
Spin-Orbital-Coupling – SOC)

Exchange-Correlation functional used to generate pseudo-potential.
Please use PBE (Perdew-Burke-Ernzerhof).
<https://journals.aps.org/prl/abstract/10.1103/PhysRevLett.77.3865>

Choose the PSP8 format

Exchange-correlation (in ABINIT)

- Kohn-Sham eigenvalues problem:

$$\left[-\frac{\hbar^2 \nabla^2}{2m_e} + U_{\text{ext}}(\mathbf{r}) + U_{\text{H}}(\mathbf{r}) + U_{\text{xc}}(\mathbf{r}) \right] \phi_{n\mathbf{k}}(\mathbf{r}) = \varepsilon_{n\mathbf{k}} \phi_{n\mathbf{k}}(\mathbf{r})$$

- $U_{\text{xc}} = \delta E_{\text{xc}} / \delta n$, with $E_{\text{xc}}[n]$: exchange-correlation functional
- ABINIT supports many types of exchange-correlation functionals (input variable **ixc**)
 - Local-Density-Approximation (LDA): $E_{\text{xc}}[n(\mathbf{r})] = \int d\mathbf{r} \varepsilon_{\text{xc}}(n(\mathbf{r})) n(\mathbf{r})$
 - Generalized-Gradient-Approximation (GGA): $E_{\text{xc}}[n(\mathbf{r})] = \int d\mathbf{r} \varepsilon_{\text{xc}}(n(\mathbf{r}), \nabla_{\mathbf{r}} n(\mathbf{r})) n(\mathbf{r})$
 - * PBE is a type of GGA
- Choice must be consistent with pseudopotentials used!
- ABINIT selects **ixc** automatically, depending on given pseudopotentials
→ no need to specify variable in input file

k-points grid

- Eigenvalue problem $\mathbf{H}_{\text{KS}}(\mathbf{k}) \mathbf{c}_{n\mathbf{k}} = \varepsilon_{n\mathbf{k}} \mathbf{c}_{n\mathbf{k}}$ can be solved only of a finite number of \mathbf{k} -points
- Properties of interest (total energy, charge density, ...) from average:

$$\overline{F} = \frac{1}{\text{card}(\mathcal{K})} \sum_{\mathbf{k} \in \mathcal{K}} F(\mathbf{k})$$

where $\mathcal{K} \subset \text{BZ}$ and $\text{card}(\mathcal{K}) < \infty$.

- Regular grid within Irreducible Brillouin Zone (IBZ):
 1. Monkhorst-Pack grid: <https://journals.aps.org/prb/abstract/10.1103/PhysRevB.13.5188>
 $\mathbf{k}_{prs} = u_p \mathbf{G}_1 + u_r \mathbf{G}_2 + u_s \mathbf{G}_3$, where $u_i = (2i - q_i - 1)/(2q_i)$, with $i = 1, \dots, q_i$ ($i = p, r, s$)
 2. Sets of \mathbf{k} -points with equal F -value because of **symmetry** replaced by one \mathbf{k} -point weighted according to degeneracy

k-points grid in ABINIT

- Inside ABINIT input file **something.abi**:

```
#####
# k-POINT GRID (see https://journals.aps.org/prb/abstract/10.1103/PhysRevB.13.5188)
#####
kptopt 1  ← algorithm to set-up k points
            • > 0: grid + symmetry (average quantities)
            • < 0: segments (band structure)

ngkpt 2 2 2  ← qp, qr, qs
nshiftk 4
shiftk 0.5 0.5 0.5
        0.5 0.0 0.0
        0.0 0.5 0.0
        0.0 0.0 0.5
```

Diagram illustrating the shiftk matrix structure. A blue bracket groups the four rows of the shiftk matrix. A blue arrow points from the text "To include origin:" to the second row of the matrix (0.5 0.0 0.0).

method to set-up k-points;
1: symmetry accounted for,
so that only k-points
within irreducible BZ generated,
with appropriate symmetry-defined weights
(degeneracy); prevents redundancy
number of points per grid
number of shifted grids to be used concurrently
k-grid shift vectors; this combination is very
efficient for FCC lattice;
other choices might be more suitable for other
lattices (see
https://docs.abinit.org/variables/basic/#shiftk)

Default:

nshiftk 1

shiftk 0.5 0.5 0.5

To include origin:

nshiftk 1

shiftk 0.0 0.0 0.0

Self-consistency

- Kohn-Sham eigenvalue problem:

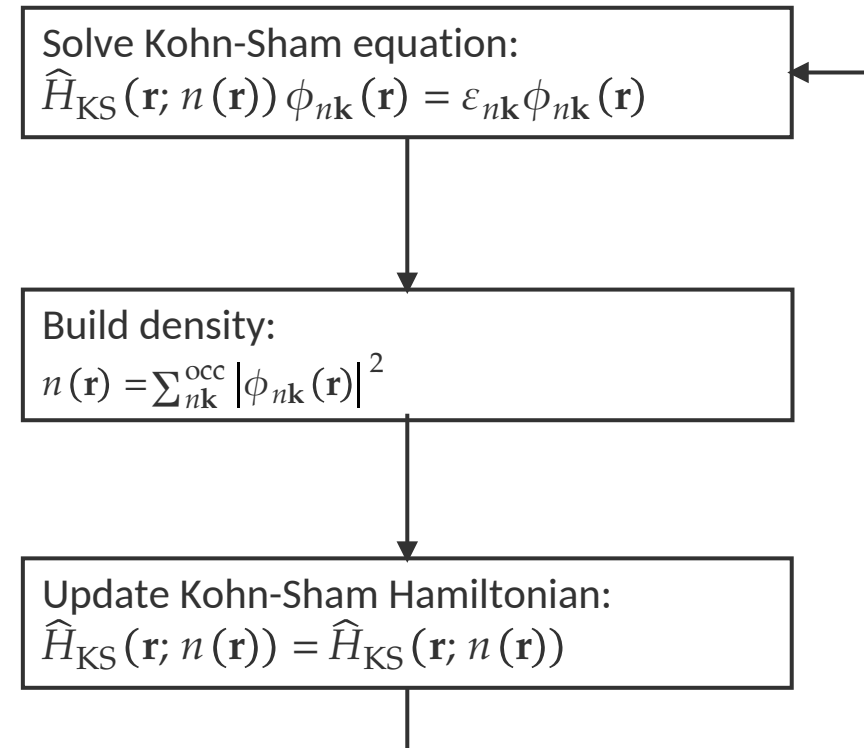
$$\left[-\frac{\hbar^2 \nabla^2}{2m_e} + U_{\text{ext}}(\mathbf{r}) + U_{\text{H}}(\mathbf{r}) + U_{\text{xc}}(\mathbf{r}) \right] \phi_{n\mathbf{k}}(\mathbf{r}) = \varepsilon_{n\mathbf{k}} \phi_{n\mathbf{k}}(\mathbf{r})$$

* specified * something missing...

- $U_{\text{H}}(\mathbf{r}) = U_{\text{H}}(\mathbf{r}; n(\mathbf{r})) = \frac{e^2}{2} \int d\mathbf{r}' \frac{n(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|}$
 $U_{\text{xc}}(\mathbf{r}) = U_{\text{xc}}(\mathbf{r}; n(\mathbf{r}))$
 → dependence on solution through

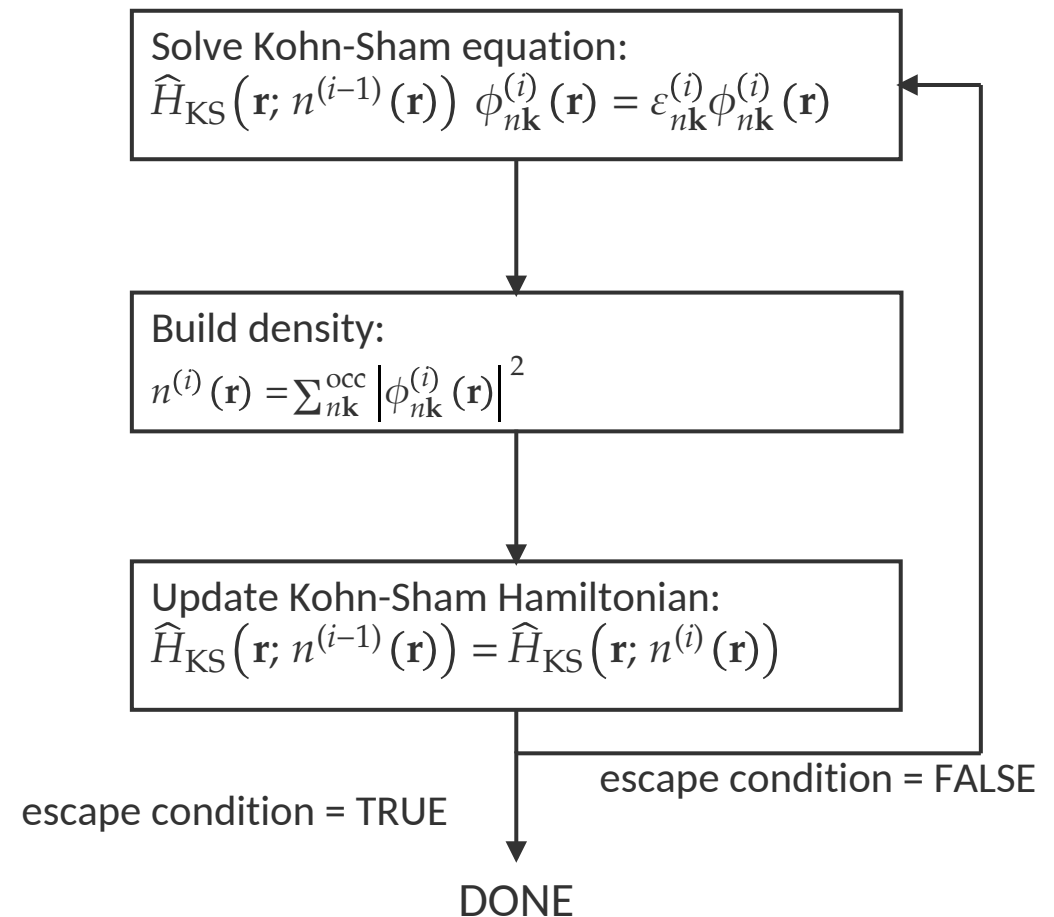
$$n(\mathbf{r}) = \sum_{n\mathbf{k}}^{\text{occ}} |\phi_{n\mathbf{k}}(\mathbf{r})|^2$$

- Solution must be self-consistent!



Self-consistency in ABINIT (1/2)

- Starting from initial guess $n^{(0)}(\mathbf{r})$, perform self-consistent cycle until escape condition fulfilled:
 - $i > \text{max. number of cycles}$
 - $\text{metric} < \text{tolerance}$
- Metric = change of selected quantity from previous step. One among:
 - toldfe**: total energy difference
 - toldff**: atomic forces difference
 - tolwfr**: wave function squared residual
 - tolvrs**: potential energy residual
- * see <https://docs.abinit.org/topics/SCFControl/>



Self-consistency in ABINIT (2/2)

- Inside ABINIT input file **something.abi**:

```
#####  
# SCF PROCEDURE  
#####  
nstep 10                                # max. number of SCF cycles  
toldfe 1.0d-6                          # convergence criterion: absolute energy difference  
                                         # between two consecutive steps
```

```
#####
# PREAMBLE
#####
outdata_prefix "./scf/scf" # path prefix for output data files

#####
# UNIT CELL
#####
acell 10.18 10.18 10.18 # [Bohr] length scales by which dimensionless primitive translation vectors 'rprim' are multiplied
rprim 0.0 0.5 0.5 # dimensionless primitive translation vectors
      0.5 0.0 0.5
      0.5 0.5 0.0

#####
# ATOM TYPES
#####
ntypat 1 # number of atom types
znuc1 14 # atomic number of the atom types (space-separated)
pp_dirpath "../.../Pseudos/PBE-SR" # path to directory containing pseudopotentials (within quotation marks)
pseudos "Si.psp8" # pseudopotential file names (within quotation marks, comma-separated)

#####
# ATOMS
#####
natom 2 # number of atoms in the unit cell
typat 1 1 # type of the atoms within the unit cell (space separated, must be consistent with 'znuc1' and 'pseudos' order)
xred 0 0 0 # atom fractional coordinates within unit cell
     1/4 1/4 1/4

#####
# PLANE-WAVE BASIS SET
#####
ecut 12.0 # [Ha] plane-wave cut-off energy (recommended values at http://www.pseudo-djo.org/)

#####
# k-POINT GRID (see https://journals.aps.org/prb/abstract/10.1103/PhysRevB.13.5188)
#####
kptopt 1 # method to set-up k-points;
          # 1: symmetry accounted for,
          # so that only k-points within irreducible BZ generated,
          # with appropriate symmetry-defined weights (degeneracy);
          # prevents redundancy
          # number of points per grid
ngkpt 2 2 2 # number of shifted grids to be used concurrently
nshiftk 4 # k-grid shift vectors; this combination is very efficient for FCC lattice;
          # other choices might be more suitable for other lattices
          # (see https://docs.abinit.org/variables/basic/#shiftk)
shiftk 0.5 0.5 0.5
      0.5 0.0 0.0
      0.0 0.5 0.0
      0.0 0.0 0.5

#####
# SCF PROCEDURE
#####
nstep 10 # max. number of SCF cycles
toldfe 1.0d-6 # convergence criterion: absolute energy difference between two consecutive steps
```

Ground-state (GS) energy in ABINIT

- Text file just discussed instructs ABINIT to execute self-consistent GS calculation
- Open a terminal, go to the directory containing the file, and run:
`abinit something.abi 1> something.log 2> something.err`
- Summary output file `something.abo` with important info saved in same directory:

```
== END DATASET(S) =====  
=====  
-outvars: echo values of variables after computation -----  
  acell      1.018000000E+01  1.018000000E+01  1.018000000E+01 Bohr  
  amu        2.80855000E+01  
  diemac     1.20000000E+01  
  ecut       1.20000000E+01 Hartree  
  etotal     -8.4553030645E+00  
  fcart      1.0580138401E-27 -6.3480830408E-28 -1.0580138401E-27  
            -1.0580138401E-27  6.3480830408E-28  1.0580138401E-27
```

total energy (Ha)

forces on unit cell atoms
(Ha/Bohr)

each row = force vector on
one atom

- Output data files stored in the same directory by default. It is “cleaner” to specify `outdata_prefix` in `something.abi`. For example, if `./scf/scf` given:



- Unless specified otherwise, ABINIT uses Hartree (Ha) for energy and Bohr for length

Matters of convergence: part 1

- Numerical computation requires finite matrices:
 - plane-wave basis set cut-off energy **ecut** specified by the user
 - number of **k**-points **ngkpt** specified by the user
- Issue: results dependent on choice of **ecut** and **ngkpt**!
- Solution:
 1. Decide tolerance for total energy change as a function of **ecut** and **ngkpt**
 2. Calculate total energy as a function of **ecut** and **ngkpt**
 3. Use smallest (fastest) value that satisfies tolerance for subsequent calculations
- Always verify convergence, even if recommended values for **ecut** provided at PseudoDojo!

Datasets in ABINIT (1/2)

- Multi-dataset mode: <https://docs.abinit.org/guide/abinit/#34-the-multi-dataset-mode>
 - redo calculations for different values of some parameter, letting all other things equal
 - chains of calculations
- Inside ABINIT input file **something.abi**:

convergence w.r.t. cut-off energy

```
#####
#PREAMBLE
#####
outdata_prefix "./etot-vs-ecut/etot-vs-ecut"
ndtset 5
#####
# PLANE-WAVE BASIS SET
#####
ecut1 8.0
ecut2 10.0
ecut3 12.0
ecut4 14.0
ecut5 16.0
```

← number of datasets

ecut values considered
appended integer identifies dataset

convergence w.r.t. number of k-points

```
#####
#PREAMBLE
#####
outdata_prefix "./etot-vs-nkpt/etot-vs-nkpt"
ndtset 4
#####
# k-POINT GRID
#####
kptopt 1
ngkpt1 2 2 2
ngkpt2 4 4 4
ngkpt3 6 6 6
ngkpt4 8 8 8
nshiftk 4
shiftk 0.5 0.5 0.5
        0.5 0.0 0.0
        0.0 0.5 0.0
        0.0 0.0 0.5
```

ngkpt values considered
appended integer identifies dataset

Datasets in ABINIT (2/2)

- Dataset-dependent results in **something.abo** + output files for each dataset:

convergence w.r.t. cut-off energy

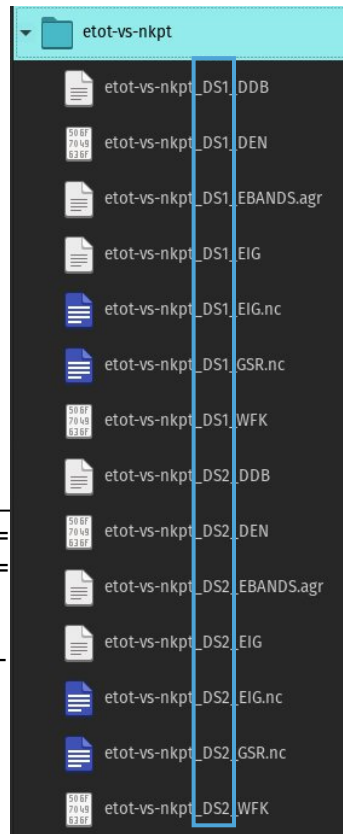
```
== END DATASET(S) =====
=====

-outvars: echo values of variables after computation -----
  acell      1.018000000E+01  1.018000000E+01  1.01
  amu        2.80855000E+01
  diemac     1.20000000E+01
  ecut1      8.00000000E+00 Hartree
  ecut2      1.00000000E+01 Hartree
  ecut3      1.20000000E+01 Hartree
  ecut4      1.40000000E+01 Hartree
  ecut5      1.60000000E+01 Hartree
  etotal1    -8.4519427804E+00
  etotal2    -8.4546405786E+00
  etotal3    -8.4553030645E+00
  etotal4    -8.4554489181E+00
  etotal5    -8.4555749123E+00
  fcart1     -1.6928221442E-27 -5.0784664327E-27  5.0784664327E-27
              1.6928221442E-27  5.0784664327E-27 -5.0784664327E-27
  fcart2     5.2900692007E-29 -5.2900692007E-29 -0.0000000000E+00
              -5.2900692007E-29  5.2900692007E-29 -0.0000000000E+00
  fcart3     1.0580138401E-27 -6.3480830408E-28 -1.0580138401E-27
              -1.0580138401E-27  6.3480830408E-28  1.0580138401E-27
  fcart4     5.2900692007E-28 -3.1740415204E-28 -1.0580138401E-28
              -5.2900692007E-28  3.1740415204E-28  1.0580138401E-28
  fcart5     -2.1160276803E-28  0.0000000000E+00  2.1160276803E-28
              2.1160276803E-28  0.0000000000E+00 -2.1160276803E-28
```

convergence w.r.t. number of k-points

```
== END DATASET(S) =====
=====

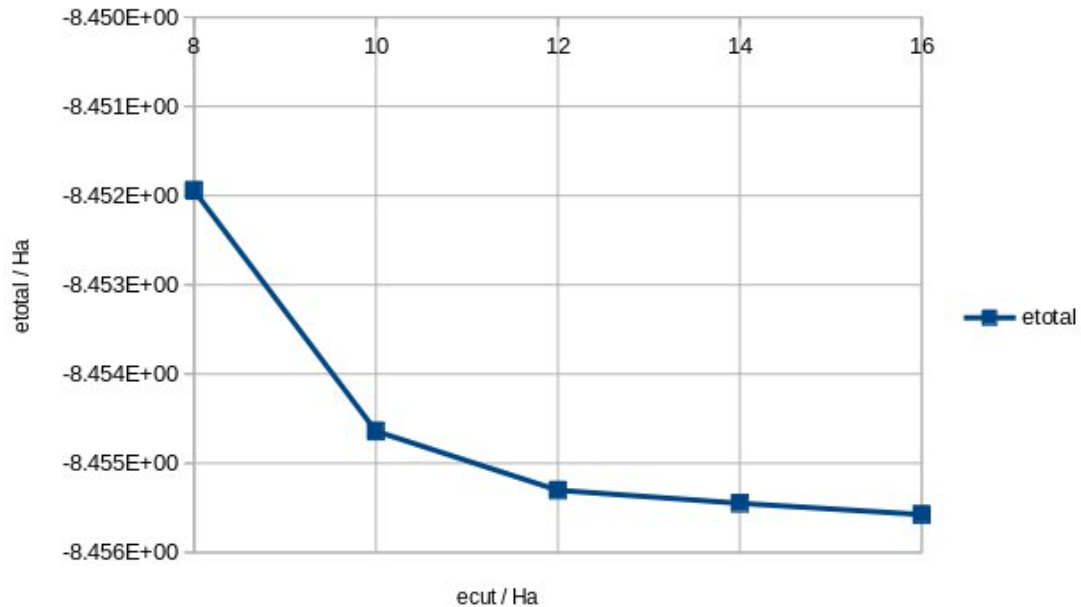
-outvars: echo values of variables after computation -----
  acell      1.018000000E+01  1.018000000E+01
  amu        2.80855000E+01
  diemac     1.20000000E+01
  ecut       1.20000000E+01 Hartree
  etotal1    -8.4553030645E+00
  etotal2    -8.4612905266E+00
  etotal3    -8.4613855371E+00
  etotal4    -8.4613885783E+00
  fcart1     1.0580138401E-27 -6.3480830408E-28 -1.0580138401E-27
              -1.0580138401E-27  6.3480830408E-28  1.0580138401E-27
  fcart2     -4.2320553606E-28  0.0000000000E+00  4.2320553606E-28
              4.2320553606E-28  0.0000000000E+00 -4.2320553606E-28
  fcart3     -0.0000000000E+00 -0.0000000000E+00 -0.0000000000E+00
              -0.0000000000E+00 -0.0000000000E+00 -0.0000000000E+00
  fcart4     0.0000000000E+00 -2.6450346003E-29  1.0580138401E-28
              0.0000000000E+00  2.6450346003E-29 -1.0580138401E-28
  kpt1      -2.50000000E-01  5.00000000E-01  0.00000000E+00
              -2.50000000E-01  0.00000000E+00  0.00000000E+00
  kpt2      -1.25000000E-01 -2.50000000E-01  0.00000000E+00
              -1.25000000E-01  5.00000000E-01  0.00000000E+00
              -2.50000000E-01 -3.75000000E-01  0.00000000E+00
              -1.25000000E-01 -3.75000000E-01  1.25000000E-0126
              -1.25000000E-01  2.50000000E-01  0.00000000E+00
```



Matters of convergence: part 2

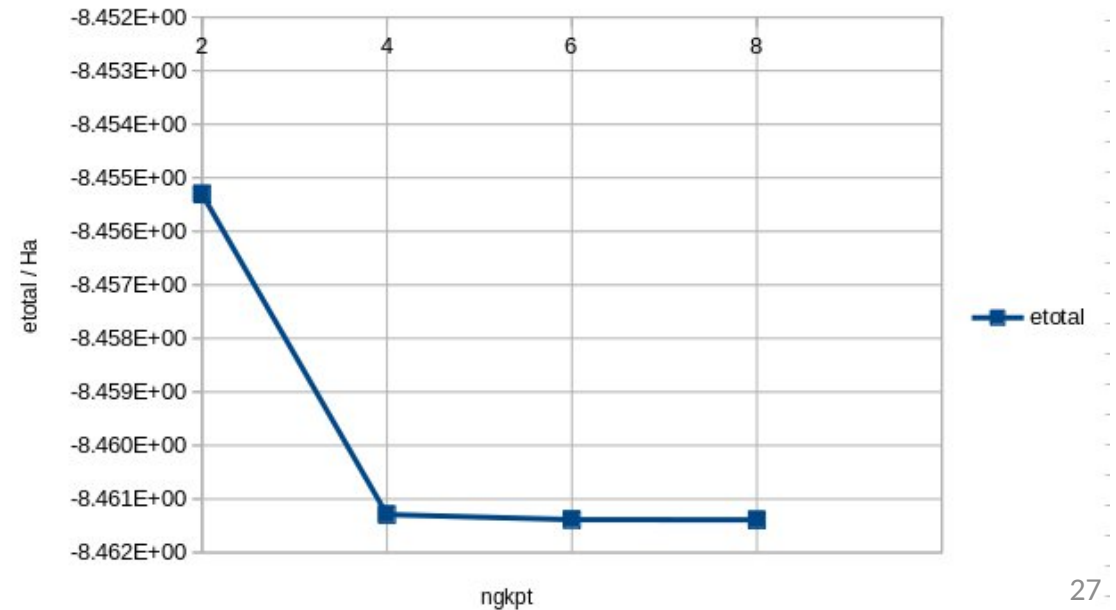
- Example. If I decide that convergence achieved when total energy change < 0.001 Ha
convergence w.r.t. cut-off energy

<u>ecut</u>	<u>etotal</u>	<u>Δetotal</u>			
8	-8.452E+00				
10	-8.455E+00	-0.003			
12	-8.455E+00	-0.001			
14	-8.455E+00	0.000			
16	-8.456E+00	0.000			



- convergence w.r.t. number of k-points

<u>ngkpt</u>	<u>etotal</u>	<u>Δetotal</u>			
2	-8.455E+00				
4	-8.461E+00	-0.006			
6	-8.461E+00	0.000			
8	-8.461E+00	0.000			



How to deal with metals

- Issue: finite BZ sampling makes convergence difficult for metals.
Ex. For \mathbf{k} -points close to Fermi surface, highest occupied bands can enter or exit averaging sums from one iteration step to the next of the self-consistent cycle \rightarrow numerical instability.
- Solution: smear Fermi surface by introducing distribution of occupation numbers.
- How:
 1. Broaden Kohn-Sham energy levels by smearing function:
$$\delta_{\sigma}(\varepsilon) = \frac{1}{\sigma} \tilde{\delta}\left(\frac{\varepsilon}{\sigma}\right), \quad \text{where } \tilde{\delta}: \text{smearing function, with } \sigma: \text{smearing parameter}$$
 2. Local density of states becomes: $n(\mathbf{r}, \varepsilon) = \sum_{n\mathbf{k}}^{\text{occ}} \frac{1}{\sigma} \tilde{\delta}\left(\frac{\varepsilon - \varepsilon_{n\mathbf{k}}}{\sigma}\right) |\phi_{n\mathbf{k}}(\mathbf{r})|^2$
 3. Electron density becomes:
$$n(\mathbf{r}) = \sum_{n\mathbf{k}}^{\text{occ}} \tilde{\theta}\left(\frac{\varepsilon_{\text{F}} - \varepsilon_{n\mathbf{k}}}{\sigma}\right) |\phi_{n\mathbf{k}}(\mathbf{r})|^2, \quad \text{with } \tilde{\theta}(x) = \int_{-\infty}^x dy \tilde{\delta}(y) \text{ and } \# \text{electrons} = \sum_{n\mathbf{k}}^{\text{occ}} \tilde{\theta}\left(\frac{\varepsilon_{\text{F}} - \varepsilon_{n\mathbf{k}}}{\sigma}\right)$$
- Advantage: local density-of-states can now be computed accurately on finite grid if energy levels separation small with respect to σ

How to deal with metals in ABINIT

- Inside ABINIT input file **something.abi**:

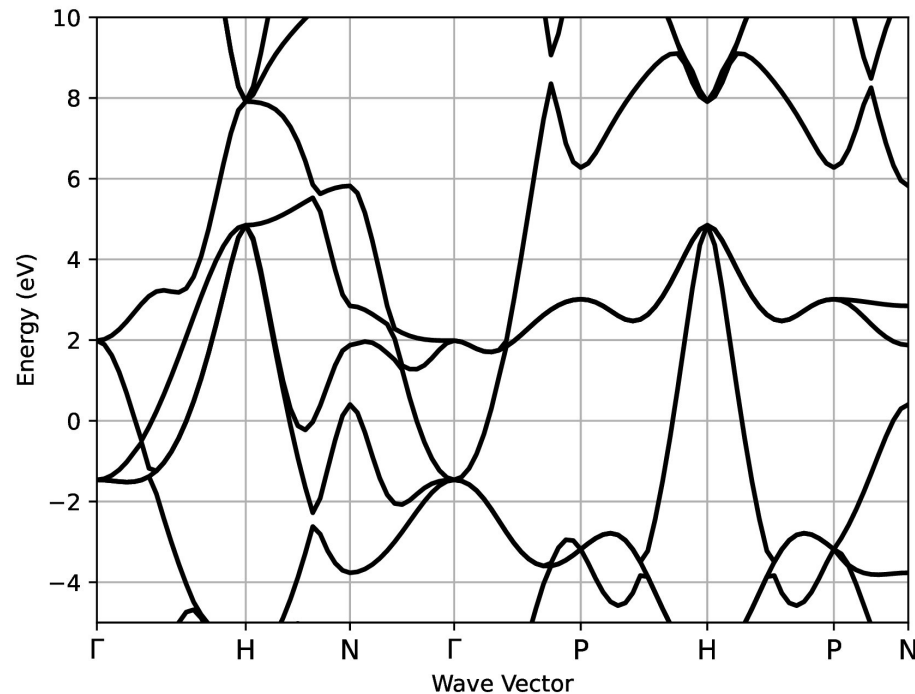
```
#####  
# OCCUPATION  
#####  
occopt 4 ← type of  $\tilde{\delta}$           # controls how occupation of electronic states is handled  
                                     # (see https://docs.abinit.org/variables/basic/#occopt)  
                                     # 4: metallic occupation with "cold smearing" of Marzari as smearing function  
                                     # smearing improves convergence w.r.t. to BZ sampling in metals,  
                                     # where state occupation drops suddenly to zero as Fermi level crossed  
                                     # otherwise, a very fine needed for accurate integration around the discontinuity  
tsmear 0.05 ←  $\sigma$  value          # [Ha] "temperature" of smearing (sigma)  
                                     # convergence should be checked also w.r.t. this parameter  
                                     # see http://theosrv1.epfl.ch/Main/ElectronicTemperature
```

- Smearing is important only for metals (e.g., aluminum) and doped semiconductors.
- You should check convergence of ground-state energy also with respect to **tsmear**, to set the appropriate value, as we have seen before for **ecut** and **ngkpt**!
- **Remark.** Do not use a too large value, otherwise the results will be converged but wrong. Suggestion: start from small (e.g., 0.001) value and slightly increase up to convergence.
- **Remark.** Convergence with respect to **ecut**, **ngkpt** and **tsmear** is not independent. If you want to be rigorous, you should check several combinations of these parameters.

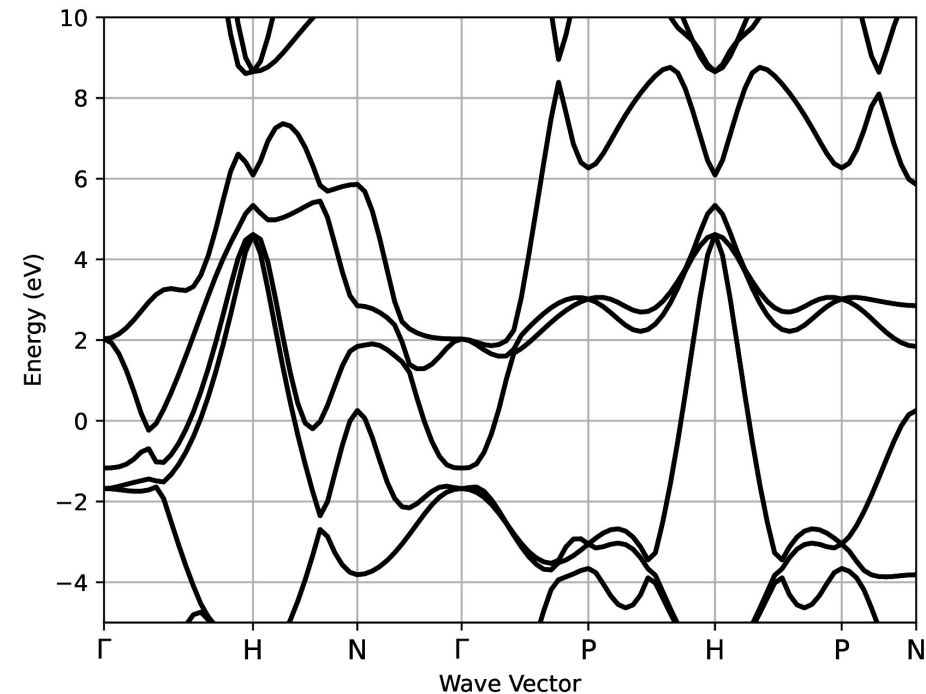
Spin-Orbit-Coupling (SOC) in ABINIT

- Inside ABINIT input file **something.abi**:
 - set **nspinor** = 2 (number of wavefunctions spinorial components – see <https://docs.abinit.org/variables/gstate/#nspinor>)
 - use Full Relativistic (FR) pseudo-potentials from PseudoDojo
 - double the value of **nband** to calculate the same number of states as for the case without SOC

W BCC - SOC off



W BCC - SOC on



Structural relaxation

Crystal structure relaxation

- Issue: crystal structure provided by the user is never the equilibrium one, i.e. atomic forces $\mathbf{F}_I = -\partial_{\mathbf{R}_I} E(\mathbf{R}) \neq 0$
- Solution: vary
 - primitive translation vectors (unit cell) $\mathbf{R}_1, \mathbf{R}_2, \mathbf{R}_3$
 - atomic coordinates within unit cell (basis vectors) $\{\mathbf{x}_\alpha\}_\alpha$until $\mathbf{F}_I = -\partial_{\mathbf{R}_I} E(\mathbf{R}) = 0$

Crystal structure relaxation in ABINIT (1/2)

- Approach:
 1. Self-consistent cycle on user-provided structure
 2. Update structure according to specified algorithms
 3. Self-consistent cycle on updated structure
 4. Repeat 2-3 until stopping criterion fulfilled
- Inside **something.abi**:

Stopping criteria:

- **ntime**: maximum number of iterations
- **tolmxf**: maximal change of atomic forces

Default: 5e-5 Ha/Bohr. Specify in input file only if you want a different value.

```
#####
# UNIT CELL RELAXATION PROCEDURE
#####
optcell 1      # optimization type;
                # 1: optimize only volume, 2: full optimization of cell geometry
                # (see https://docs.abinit.org/variables/rlx/#optcell)
ionmov  2      # algorithm to change displacement of atoms;
                # 0: fixed, 1: molecular dynamics with viscous damping,
                # 2: Broyden-Fletcher-Goldfarb-Shanno minimization (BFGS)
                # (see https://docs.abinit.org/variables/rlx/#ionmov)
ntime  10      # maximum number of relaxation steps
dilatmx 1.05   # maximum allowed dilatation of unit cell
                # needed to book additional memory:
                # increasing lattice size decreases reciprocal primitive vectors lengths,
                # which lead to more plane waves at constant energy cut-off
                # [Ha] smearing needed to remove "numerical" discontinuities
                # introduced by change of number of plane waves when unit cell size changes
                # (see https://journals.aps.org/prb/abstract/10.1103/PhysRevB.93.205147)

ecutsm  0.5
```

optimization algorithm
for cell vectors

optimization algorithm
for basis vectors

```
#####
# SCF PROCEDURE
#####
nstep 10
toldfe 1.0d-6
getwfk -1
```

Initial guess from previous
relaxation step to speed up
computation

Crystal structure relaxation in ABINIT (2/2)

- Step-dependent results in **something.abo**:

```
--- Iteration: ( 3/10) Internal Cycle: (1/1) ← relaxation step
```

```
-----SELF-CONSISTENT-FIELD CONVERGENCE-----
```

```
--- !BeginCycle
```

```
iteration_state: {dtset: 1, itime: 3, icycle: 1, }
```

```
solver: {iscf: 7, nstep: 10, nline: 4, wfoptalg: 0, }
```

```
tolerances: {toldfe: 1.00E-06, }
```

```
...
```

	iter	Etot(hartree)	deltaE(h)	residm	vres2
ETOT	1	-8.4620285139361	-8.462E+00	1.717E-06	2.899E-01
ETOT	2	-8.4621521640577	-1.237E-04	3.473E-07	1.766E-02
ETOT	3	-8.4621572711599	-5.107E-06	3.417E-07	5.050E-04
ETOT	4	-8.4621573579434	-8.678E-08	7.893E-08	1.988E-06
ETOT	5	-8.4621573582186	-2.752E-10	8.103E-09	1.330E-08

self-consistent cycle
within relaxation step

At SCF step 5, etot is converged :

for the second time, diff in etot= 2.752E-10 < toldfe= 1.000E-06

- Equilibrium structure at bottom in **something.abo**:

```
== END DATASET(S) =====
```

```
=====
```

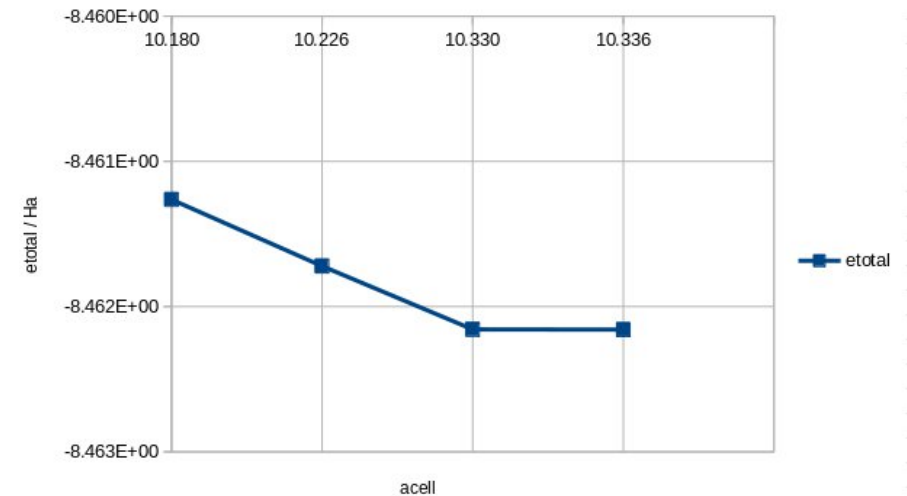
```
-outvars: echo values of variables after computation -----
```

acell	1.0335899940E+01	1.0335899940E+01	1.0335899940E+01	Bohr
amu	2.80855000E+01			
diemac	1.20000000E+01			
dilatmx	1.05000000E+00			
ecut	1.20000000E+01	Hartree		
ecutsm	5.00000000E-01	Hartree		
etotal	-8.4621588530E+00			
fcart	2.0841109057E-28	-0.0000000000E+00	-2.0841109057E-28	
	-2.0841109057E-28	-0.0000000000E+00	2.0841109057E-28	

L_i

F_I

acell / Bohr	etotal	Δetotal			
10.180	-8.461E+00				
10.226	-8.462E+00	-4.59E-04			
10.330	-8.462E+00	-4.36E-04			
10.336	-8.462E+00	-1.49E-06			



R_i/L_i

rprim	0.0000000000E+00	5.0000000000E-01	5.0000000000E-01
	5.0000000000E-01	0.0000000000E+00	5.0000000000E-01
	5.0000000000E-01	5.0000000000E-01	0.0000000000E+00

x_α

xred	-6.7047150131E-28	3.1200116404E-28	3.3657634175E-28
	2.5000000000E-01	2.5000000000E-01	2.5000000000E-01

Electronic properties

Band structure (Kohn-Sham)

- Goal: obtain ε vs $n\mathbf{k}$ (including empty states) along selected high-symmetry lines of the BZ by solving eigenvalue equation $\hat{H}_{KS} \phi_{n\mathbf{k}}(\mathbf{r}) = \varepsilon_{n\mathbf{k}} \phi_{n\mathbf{k}}(\mathbf{r})$
- Catch: ground-state calculation of electron density and total energy only need occupied states and a “coarse” grid of k-points to converge
- Approach:
 1. Ground-state calculation without (redundant) empty states that would only slow down the self-consistent cycle and “coarse” grid of k-points
 2. Non-self-consistent calculation including empty states on a fine \mathbf{k} -sampling of the high-symmetry lines with electron density from previous self-consistent step

Band structure (Kohn-Sham) in ABINIT (1/4)

- Inside ABINIT input file **something.abi**:

```
#####
# PREAMBLE
#####
outdata_prefix "./estruct/estruct"  # path prefix for output data files
ndtset 2  # number of datasets
# 1: SCF procedure to get GS electron density
# 2: one for band structure calculation
```

remember: you can use multi-dataset mode to chain calculations

Dataset 1: self-consistent GS calculation of electron density

```
#####
# k-POINT GRID (see https://journals.aps.org/prb/abstract/10.1103/PhysRevB.13.5188) for SCF procedure (dataset 1)
#####
kptopt1 1  # method to set-up k-points; 1: symmetry accounted for,
# so that only k-points within irreducible BZ generated,
# with appropriate symmetry-defined weights (degeneracy);
# prevents redundancy
# number of grid points
# number of shifted grids to be used concurrently
# k-grid shift vectors;
# this combination is very efficient for FCC lattice
# (see https://docs.abinit.org/variables/basic/#shiftk)

ngkpt1 4 4 4
nshiftk1 4
shiftk1 0.5 0.5 0.5
        0.5 0.0 0.0
        0.0 0.5 0.0
        0.0 0.0 0.5

#####
# SCF PROCEDURE
#####
nstep1 20  # max. number of SCF cycles
tolldf1 1.0d-6  # convergence criterion:
# absolute energy difference between two consecutive steps
# save density from SCF procedure (dataset1),
# so that it can be used in the non-self-consistent
# calculation of band structure (dataset 2)

prtden1 1
```

k-grid for dataset 1
(GS calculation)

save electron-density from dataset 1 so that can be used for dataset 2

Band structure (Kohn-Sham) in ABINIT (2/4)

Dataset 2: non-self-consistent calculation of KS band structure

```
#####
# BAND STRUCTURE
#####
iscf2 -2  ← non-self-consistent calculation      # self-consistency of the calculation;
                                                    # 0: SCF cycle (default),
                                                    # -2: non-SCF calculation

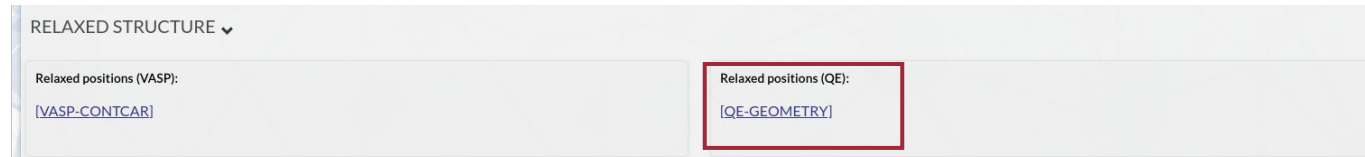
getden2 -1  ← density from previous step          # get density from previous step
kptopt2 -6  ← negative number: k-segments          # when negative, k-points set up
                                                    # from bounds and number of sub-divisions
                                                    # of segments in BZ;
                                                    # absolute value equal to the number of segments

nband2 8    ← number of states per k-point        # number of bands
              (double the value if SOC on)         # (if unsure, look at 'nband' in output file of SCF cycle
                                                    # and add some more to include empty states)

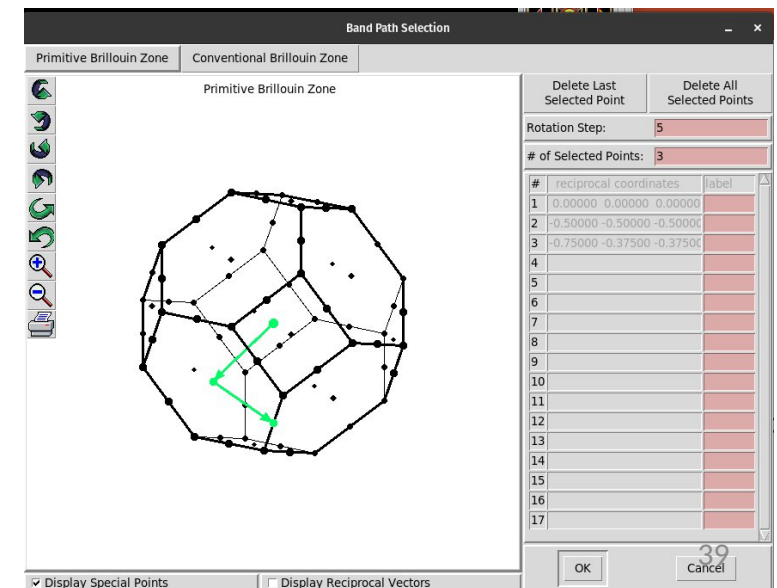
ndivsm2 10   ← number of points per segment        # number of sub-divisions of each segment
kptbounds2 0 0 0                                     # bounds of band structure segments. Gamma point
              1/2 0 1/2                               # X point
              1/2 1/4 3/4                             # W point
              1/2 1/2 1/2                             # L point
              0 0 0                                    # Gamma point
              3/8 3/8 3/4                             # K point
              1/2 0 1/2                               # X point
                                                    # convergence criterion: tolerance on wavefunction squared residual
tolwfr2 1.0d-12
enunit2 1    ← energies in eV instead of Ha        # output energies in eV instead of Ha
```

High-symmetry points specification

- A list of the high-symmetry points associated to each type of Bravais lattice can be found in the article: <https://doi.org/10.1016/j.commatsci.2010.05.010>
- Alternatively, you can:
 - Download the crystal structure QE file from AFLOW:



- Open the file with XCrysDen: `xcrysden --pwi CONTCAR.relax.qe`
- Go to **Tools > k-Path selection**
- Select your k-path by clicking on the available high-symmetry points
- Copy the reciprocal coordinates into your ABINIT input file



Recap: input for GS calculation followed by band-structure of c-Si

```
#####
# PREAMBLE
#####
outdata_prefix "./estruct/estruct"
ndtset 2

#####
# UNIT CELL
#####
acell 10.336 10.336 10.336
rprim 0.0 0.5 0.5
      0.5 0.0 0.5
      0.5 0.5 0.0

#####
# ATOM TYPE
#####
ntypat 1
znuc1 14
pp_dirpath "../.../Pseudos/PBE-SR"
pseudos "Si.psp8"

#####
# ATOMS
#####
natom 2
typat 1 1 1
xred 0 0 0
     1/4 1/4 1/4

#####
# PLANE-WAVE BASIS SET
#####
ecut 12.0
```

```
#####
# k-POINT GRID
#####
kptopt1 1

ngkpt1 4 4 4
nshiftk1 4
shiftk1 0.5 0.5 0.5
        0.5 0.0 0.0
        0.0 0.5 0.0
        0.0 0.0 0.5

#####
# SCF PROCEDURE
#####
nstep1 20
tolwfe1 1.0d-6

prtden1 1
```

```
#####
# BAND STRUCTURE
#####
iscf2 -2

getden2 -1
kptopt2 -6

nband2 8

ndivsm2 10
kptbounds2 0 0 0
           1/2 0 1/2
           1/2 1/4 3/4
           1/2 1/2 1/2
           0 0 0
           3/8 3/8 3/4
           1/2 0 1/2
tolwfr2 1.0d-12
enunit2 1
```

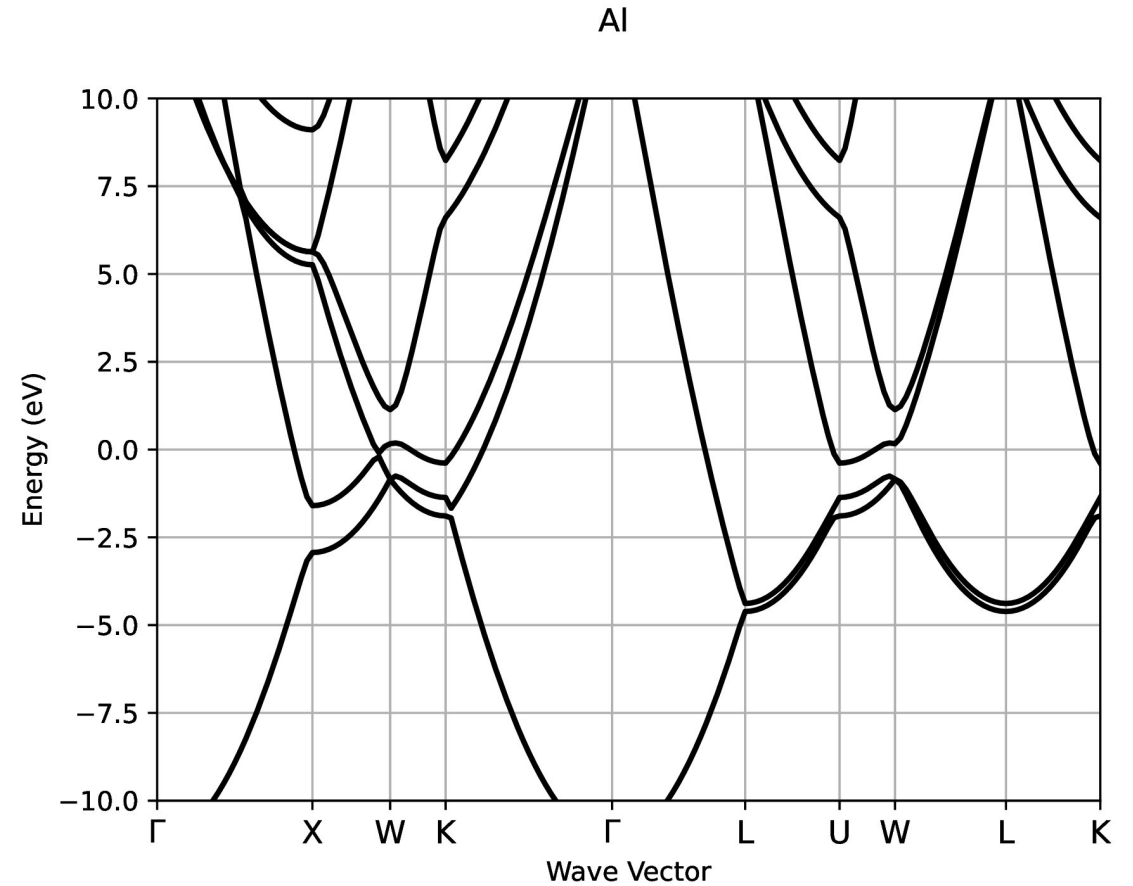
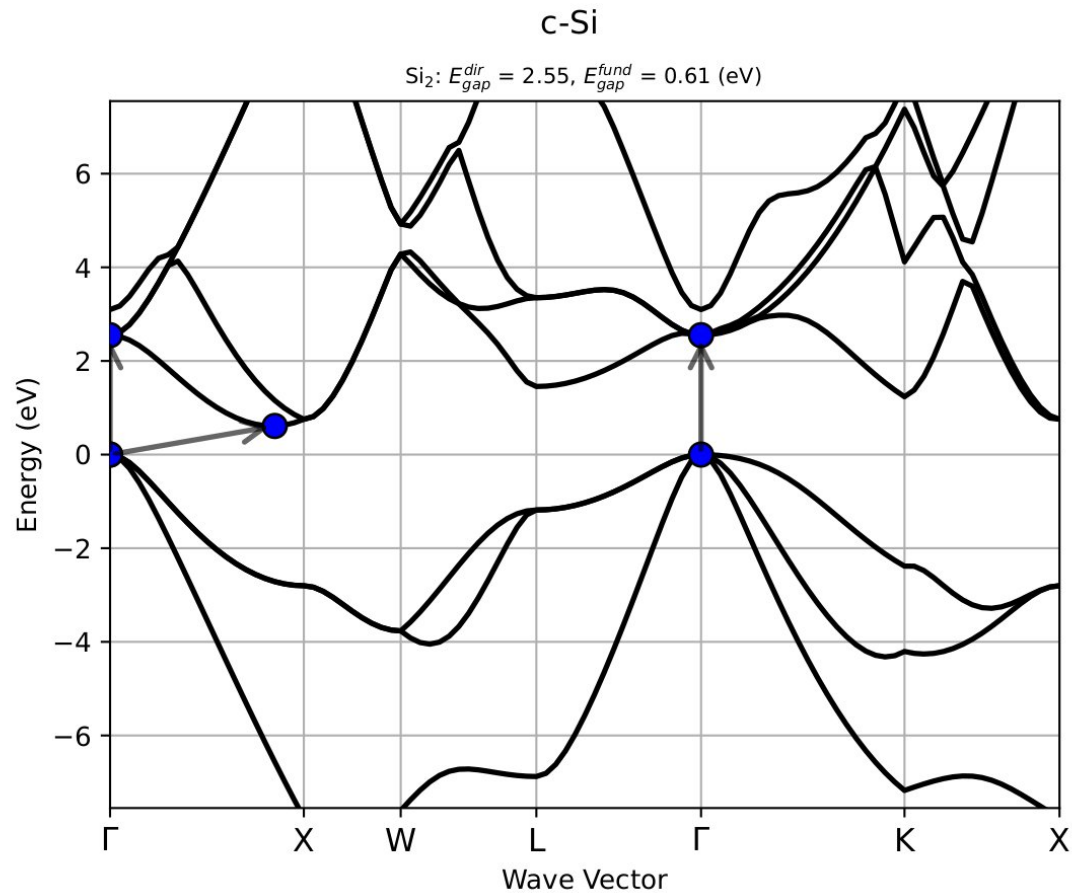

Band structure (Kohn-Sham) in ABINIT (3/4)

- Electronic states can be found inside ABINIT output file **something.abo**, under under "Eigenvalues (eV) for nkpt=[...]":

```
Eigenvalues ( eV ) for nkpt= 99 k points:
kpt#  1, nband= 8, wtk= 1.00000, kpt= 0.0000 0.0000 0.0000 (reduced coord)
-7.65220  4.16904  4.16904  4.16904  6.71902  6.71902  6.71902  7.26628
kpt#  2, nband= 8, wtk= 1.00000, kpt= 0.0250 0.0000 0.0250 (reduced coord)
-7.64146  4.09112  4.12327  4.12327  6.69248  6.77808  6.77808  7.34253
kpt#  3, nband= 8, wtk= 1.00000, kpt= 0.0500 0.0000 0.0500 (reduced coord)
-7.60924  3.87793  3.99753  3.99753  6.61599  6.94365  6.94365  7.54778
kpt#  4, nband= 8, wtk= 1.00000, kpt= 0.0750 0.0000 0.0750 (reduced coord)
-7.55560  3.57129  3.81680  3.81680  6.49775  7.19015  7.19015  7.82965
kpt#  5, nband= 8, wtk= 1.00000, kpt= 0.1000 0.0000 0.1000 (reduced coord)
-7.48061  3.20738  3.60509  3.60509  6.34869  7.49388  7.49388  8.12688
kpt#  6, nband= 8, wtk= 1.00000, kpt= 0.1250 0.0000 0.1250 (reduced coord)
-7.38437  2.80964  3.37936  3.37936  6.17975  7.83734  7.83734  8.36244
```

- The same information is stored in the ***_DS2_EBANDS.agr** output data file, that can be plotted directly with **xmgrace** (<http://plasma-gate.weizmann.ac.il/Grace/>)

Band structure (Kohn-Sham) in ABINIT (4/4)



Density-Of-States (Kohn-Sham)

- Goal: obtain Density-Of-States (DOS) vs ε (including empty states) by solving eigenvalue equation $\hat{H}_{KS} \phi_{n\mathbf{k}}(\mathbf{r}) = \varepsilon_{n\mathbf{k}} \phi_{n\mathbf{k}}(\mathbf{r})$
- Catch: ground-state calculation of electron density and total energy only need occupied states and a “coarse” grid of k-points to converge, but we are also interested in empty states and need a fine grid to get reasonable DOS values!
- Approach:
 1. Ground-state calculation without (redundant) empty states that would only slow down the self-consistent cycle and “coarse” grid of k-points
 2. Non-self-consistent calculation including empty states on a fine grid of the Brillouin Zone

Density-Of-States (Kohn-Sham) in ABINIT (1/3)

- The approach is the same as the one used for the band structure: a self-consistent GS calculation on a “coarse” BZ grid (dataset 1) followed by a non-self-consistent calculation on a much finer grid and including empty states (dataset 2)
- Inside ABINIT input file **something.abi**, the following settings are introduced for dataset 2:

```
#####  
# DENSITY-OF-STATES AND FERMI SURFACE (DATASET 2)  
#####
```

```
ngkpt2 30 30 30  
nshiftk2 1  
shiftk2 0.0 0.0 0.0  
iscf2 -3
```

← -3 instead of -2: non-self-consistent calculation
+ initialize occupation of energy levels

```
getden2 -1  
tolwfr2 1.0d-12  
nband2 6
```

← electron density from self-consistent GS calculation

```
enunit2 1  
prtdos2 2
```

← save density-of-states

```
prtfsurf2 1
```

← save Fermi surface

```
# number of grid points for DOS calculation (dataset 2);  
# many more points needed to have a fine sampling of BZ;  
  
# self-consistency of the calculation;  
# 0: SCF cycle,  
# -2: non-SCF calculation  
# (electron density from previous dataset - SCF cycle);  
# -3: like -2, but initialize occupation  
# (see https://docs.abinit.org/variables/basic/#iscf)  
# get density from previous step  
# tolerance on wavefunction squared residual  
# number of bands (if unsure, look at 'nband'  
# in output file of SCF cycle and add some more)  
# output energies in eV instead of Ha  
# save density-of-states;  
# 2: use the tetrahedron method to calculate it  
# (see https://docs.abinit.org/variables/files/#prtdos)  
# save data for Fermi surface plot in BXSf format  
# (readable by XCrysDen)
```

Recap: input for GS calculation followed by density-of-states and Fermi surface of aluminum

```
#####
# PREAMBLE
#####
outdata_prefix "./edos-fermi/edos-fermi"
ndtset 2

#####
# UNIT CELL
#####
acell 7.626 7.626 7.626
rprim 0.0 0.5 0.5
      0.5 0.0 0.5
      0.5 0.5 0.0

#####
# ATOM TYPES
#####
ntypat 1
znuc1 13
pp_dirpath "../.../Pseudos/PBE-SR"
pseudos "Al.psp8"

#####
# ATOMS
#####
natom 1
typat 1
xred 0.0 0.0 0.0

#####
# PLANE-WAVE BASIS SET
#####
ecut 14.0
```

```
#####
# OCCUPATION
#####
occopt 4
```

smearing settings
(Al is a metal!)

```
tsmear 0.04
```

```
#####
# k-POINT GRID (DATASET 1)
#####
```

```
kptopt 1 ← same for both datasets
ngkpt1 6 6 6      (no integer appended)
nshiftk1 4
shiftk1 0.5 0.5 0.5
        0.5 0.0 0.0
        0.0 0.5 0.0
        0.0 0.0 0.5
```

```
#####
# SCF PROCEDURE (DATASET 1)
#####
nstep1 20
tolDFE1 1.0d-6
diemac1 1e6
prtden1 1
```

save electron-density from dataset
1 so that can be used for dataset 2

```
#####
# DENSITY-OF-STATES AND FERMI SURFACE
# (DATASET 2)
#####
ngkpt2 30 30 30
nshiftk2 1
shiftk2 0.0 0.0 0.0
iscf2 -3
```

```
getden2 -1
tolwfr2 1.0d-12
nband2 6
```

```
enunit2 1
prtdos2 2
```

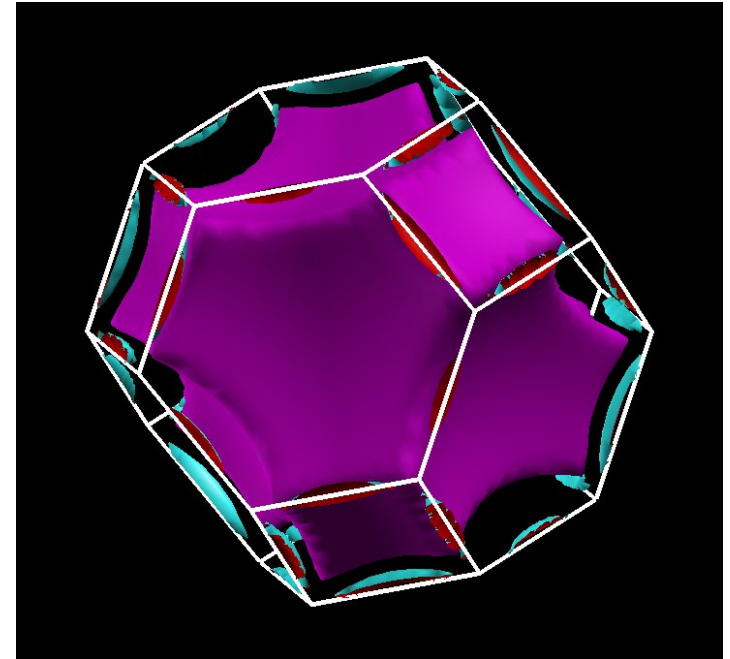
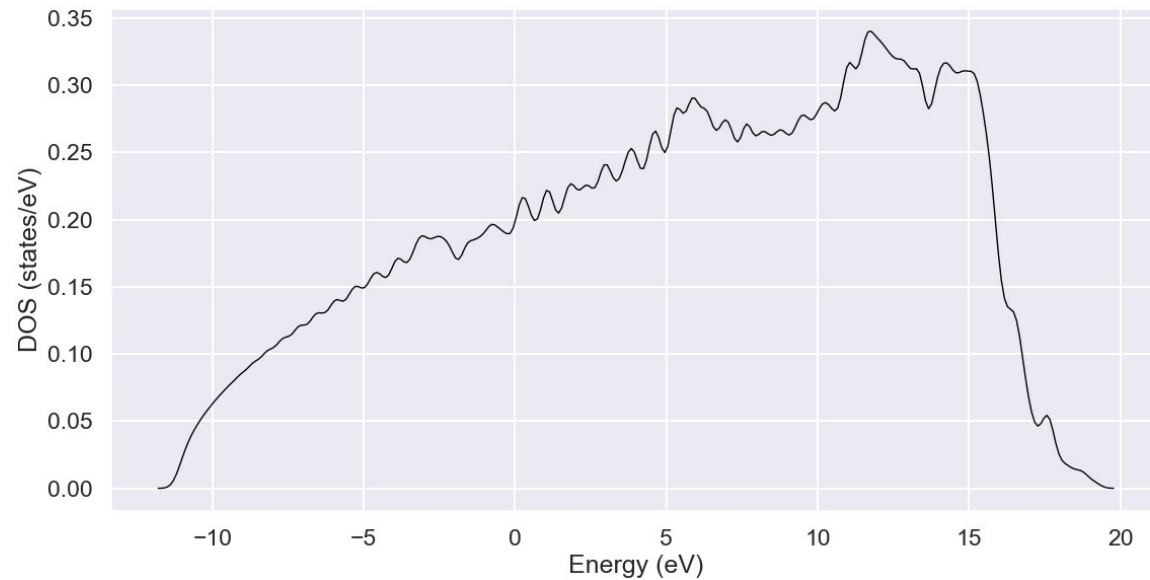
```
prtfsurf2 1
```

Density-Of-States (Kohn-Sham) in ABINIT (2/3)

- If `prtdos=2`, `*_DS2_DOS` output data file generated, which contains density-of-states calculated with the tetrahedron method and can be plot with the command `xmgrace *_DS2_DOS`
- If `prtfsurf=1`, `*_DS2_BXSF` output data file generated, which contains Fermi surface and can be directly plotted with XCrysDen () with the command `xcrysden --bxsf *_DS2_BXSF`

Density-Of-States (Kohn-Sham) in ABINIT (3/3)

- Aluminum:



Doping in ABINIT

- If you want to add or remove electrons, introduce the variable `cellcharge` in your input file.
- `cellcharge` specifies the unit cell charge in number of elementary charges.
- For example:
 - `cellcharge 1` means one electron missing per unit cell.
 - `cellcharge -1` means one electron added per unit cell.
- `cellcharge` specifies the average value. Also rational numbers can be used, meaning that one electron is added/removed only from a subset of the unit cells.

Phonon properties

Phonons with DFPT

- Phonon eigenvalues (ω) computed by solving secular equation:

$$\det \left| \frac{1}{\sqrt{M_s M_t}} C_{st}^{\alpha\beta}(\mathbf{q}) - \omega^2(\mathbf{q}) \right| = 0$$

- Dynamical matrix given by:

$$C_{st}^{\alpha\beta}(\mathbf{q}) = \frac{1}{N_c} \frac{\partial^2 E}{\partial u_s^*(\mathbf{q}) \partial u_t(\mathbf{q})}$$

- 2nd derivatives related to electron density by:
- **Linear response** can be obtained by Density-Functional-Perturbation-Theory (DFPT)


Phonons with DFPT in ABINIT (1/8)

- Approach:
 1. Ground-state calculation (dataset 1) to get electron density $n(\mathbf{r})$
 2. Calculation of linear response and dynamical matrices on a grid of \mathbf{q} -vectors
 - ABINIT expects one and only one \mathbf{q} -vector per dataset
 - \mathbf{q} -vectors must be a subset of the k-vectors grid used in the GS calculation
 3. ABINIT will create a **DDB** (Derivative Data Base) file per \mathbf{q} -vector that must be merged into a single file with command line utility **MRGDDB**
 4. Interatomic-Force-Constants (IFC) $C_{st}^{\alpha\beta}(\mathbf{R}) = \frac{1}{N_c} \sum_{\mathbf{q}} C_{st}^{\alpha\beta}(\mathbf{q}) e^{i\mathbf{q}\cdot\mathbf{R}}$ obtained with command line utility **ANADDDB**
 5. Phonon eigenvalues for arbitrary \mathbf{q} -vectors (e.g., phonon band structure and DOS) obtained by Fourier interpolation with command line utility **ANADDDB**

Phonons with DFPT in ABINIT (2/8)

- Steps 1-2: In ABINIT input file `response.abi`:


```
#####  
# PREAMBLE  
#####  
outdata_prefix "./1-response/response"      # path prefix for output data files  
ndtset 9                                     # number of datasets:  
                                              # 1: initial GS SCF calculation, 2-9: response to phonons  
                                              # the number of irreducible q-points has been checked  
                                              # with previous dummy run
```



- ABINIT will only retain the irreducible q-points, i.e. a subset of the q-point grid specified by the user
- However, number of datasets must be 1 (GS calculation) + number of q-points (1 dataset per q point)
- How to know `ndtset` before-hand?
- Do a dummy GS calculation with k-grid = desired q-grid and read the number of irreducible k-points in the output `*.abo` file

8 points

```
----- Echo of variables that govern the present computation -----  
...  
-outvars: echo values of preprocessed input variables -----  
          acell      7.626000000E+00  7.626000000E+00  7.626000000E+00 Bohr  
          amu        2.69815390E+01  
          ecut       1.40000000E+01 Hartree  
...  
nkpt      8
```



Phonons with DFPT in ABINIT (3/8)

```
#####
# PHONON-RELATED DEFAULT SETTINGS  used for each dataset (no appended integer) unless overridden
#####
qptopt 1          # q-point grid generation
ngqpt 4 4 4       # same meanings as k-point related vars
nshiftq 1
shiftq 0 0 0
nqpt 1 ← read q-points? 0: no, 1: yes
rfphon 1 ← calculate linear response? 0: no, 1: yes
getwfk 1 ← get wavefunction from dataset 1 (GS calculation)
tolvrs 1.0d-8
kptopt 3 ← k-points in full BZ, not just IBZ (needed for DFPT)

# specify which q-point from the grid to use for each dataset,
# by index in the list of q-points
iqpt: 0           # start q-point: iqpt=0 => q = 0 0 0
iqpt+ 1           # increment iqpt by 1 at each dataset
                  # iqpt=1 => q = 0 0 0
                  # at last dataset (9), iqpt=8 (max. index)
```

ABINIT accepts only ONE q-point per dataset. We need to tell ABINIT which q-point to read from the generated grid at each dataset. **iqpt**=i with i>0 means “read the i-th q-point”. **iqpt**=0 means q=(0,0,0). **iqpt: 0 iqpt+ 1** means start from 0 and increase value by 1 at each dataset. By starting from 0, we do not read from the q-point grid for dataset 1 (GS calculation), and we read all q-points one-by-one for the next datasets (see <https://docs.abinit.org/guide/abinit/#34-the-multi-dataset-mode>)

Phonons with DFPT in ABINIT (4/8)

```
#####
# DATASET 1: GROUND-STATE CALCULATION - OVERRIDE DEFAULT SETTINGS ABOVE AND BELOW
#####
getwfk1 0 # no wavefunction from previous calculation
nqpt1 0 # do not read q-point
rfphon1 0 # do not apply phonon perturbation
tolvrs1 1.0d-18 # SCF convergence criterion: residual of the potential
# (see https://docs.abinit.org/variables/basic/#tolvrs)
kptopt1 1 # override set-up of k-points to take symmetry into account
# OK for GS, must be 3 for response calculation
```

Disable phonon-specific settings for GS calculation (dataset 1)

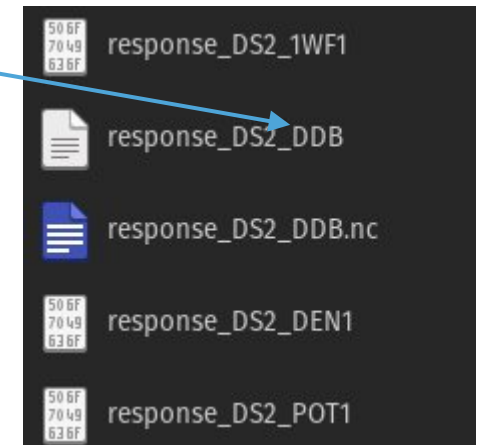
Everything else stays the same.

- One Derivative Data Base (*DDB) output data file generated per dataset, containing 2nd derivatives at the bottom:

```
2nd derivatives (non-stat.) - # elements : 21
qpt 0.00000000E+00 0.00000000E+00 0.00000000E+00 1.0
1 1 1 1 0.25461565428637D+00 0.00000000000000D+00
2 1 1 1 0.12730782714318D+00 0.00000000000000D+00
3 1 1 1 0.12730782714318D+00 0.00000000000000D+00
2 3 1 1 0.00000000000000D+00 0.00000000000000D+00
3 3 1 1 0.00000000000000D+00 0.00000000000000D+00
```

Perturbations
multi-index

Real and imaginary part
of the 2nd derivative



Phonons with DFPT in ABINIT (5/8)

- Step 3: Merge the DDB files into a single database by running the command `mrgddb < mrgddb.abi`. Important: `<` is used because `mrgddb.abi` is not really an input file that is parsed by `mrgddb`. `mrgddb` is interactive: it asks a question and wait for the user input. (try to simply run `mrgddb` without anything after). By using `<`, we supply each line of the `abi` file as “answer” to the expected `mrgddb` “question”. Therefore, the order of the lines must be exactly as below, and no comments are allowed, since they would be interpreted as input. They are only shown below to describe the `abi` file.

```
2-mrgddb.abo                                     # merged DDB filename
A1 responses                                     # description
8                                                  # number of DDBs to be merged
1-response/response_DS3_DDB                      # path to DDBs to be merged
1-response/response_DS4_DDB
1-response/response_DS5_DDB
1-response/response_DS6_DDB
1-response/response_DS7_DDB
1-response/response_DS8_DDB
1-response/response_DS9_DDB
1-response/response_DS10_DDB
```

Phonons with DFPT in ABINIT (6/8)

- Step 4-5: Calculate IFCs and interpolate phonon band structure and DOS with the command **anaddb anaddb.abi**, with the ANADDB input file **anaddb.abi** written as below:

```
#####
# PREAMBLE
#####
output_file = "3-anaddb.abo"          # output file
ddb_filepath = "2-mrgddb.abo"         # path to Derivative DataBase (DDB)
gkk_filepath = "dummy1"               # path to GKK matrix elements database (only for e-ph)
eph_prefix = "dummy2"                 # prefix for e-ph output data (only for e-ph)
ddk_filepath = "dummy3"               # not used here, but must be written
outdata_prefix = "./3-anaddb/anaddb" # output data prefix

#####
# FLAGS
#####
ifcflag 1      Activate IFC Fourier interpolation # calculate interatomic force constants and
                                                       # use them to interpolate phonon spectrum
                                                       # and dynamical matrices at every q-vector

#####
# WAVE VECTOR (COARSE) GRID FROM CALCULATED DDB BASED ON WHICH IFCs CALCULATED
#####
ngqpt 4 4 4      (sub-)grid of DDB q-vectors based on # Monkhorst-Pack grid to consider
                  which IFCs are calculated           # (must correspond to DDB data or a subset of it)
nqshft 1          # number of q-shifts
q1shft 0 0 0     # shift vectors of coarse grid
```


Phonons with DFPT in ABINIT (7/8)

```
#####
# BAND STRUCTURE CALCULATION BY FOURIER INTERPOLATION ON Q-VECTOR SEGMENTS
#####
nqpath 9                                # number of interpolation q-segment end-points
ndivsm 10                              # meaning of variables analogous to k-paths
                                           # same segment not allowed twice (change cell)

qpath 0.0000  0.0000  0.0000           # Gamma point
      0.3750  0.3750  0.7500           # K point
      0.5000  0.5000  1.0000           # X point
      1.0000  1.0000  1.0000           # Gamma point
      0.5000  0.5000  0.5000           # L point
      0.5000  0.0000  0.5000           # X point
      0.5000  0.2500  0.7500           # W point
      0.5000  0.5000  0.5000           # L point
      0.0000  0.0000  0.0000           # Gamma point

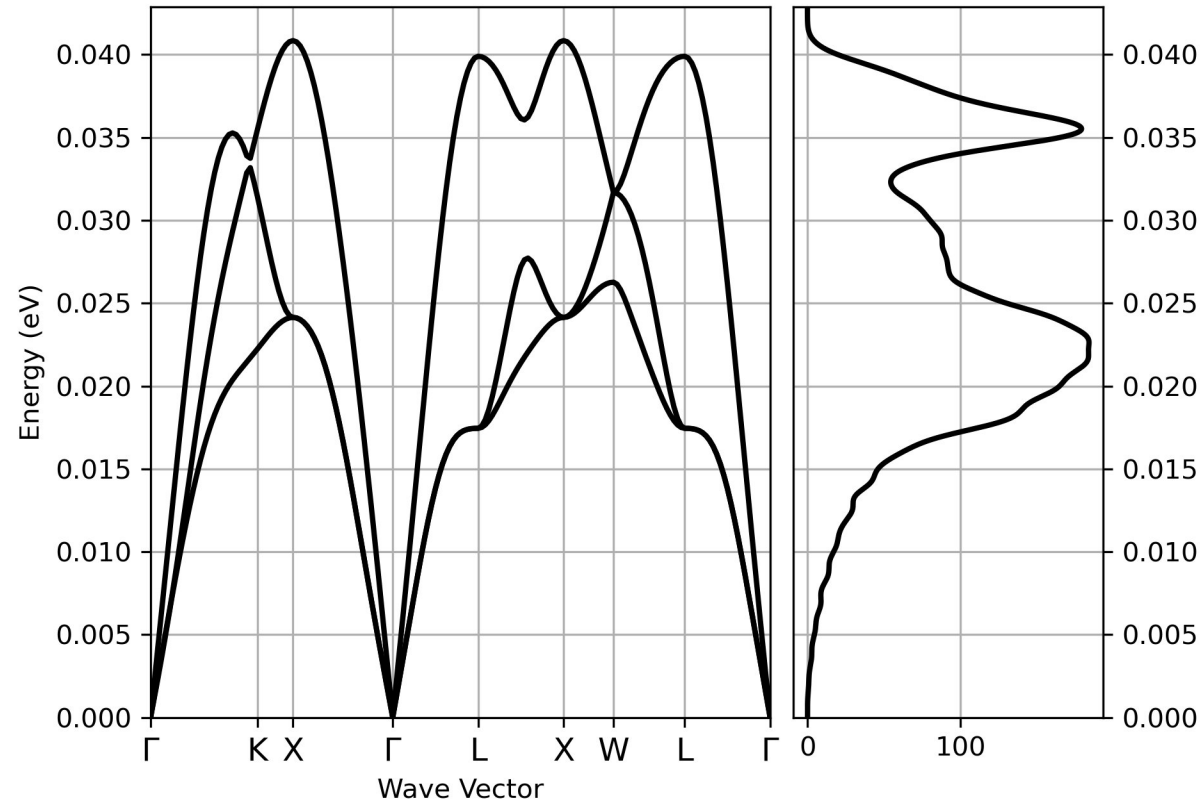
#####
# DOS CALCULATION BY FOURIER INTERPOLATION ON FINE GRID
#####
prtdos 1                                # calculate DOS?
                                           # 0: no
                                           # 1: by gaussian method
                                           # 2: by tetrahedron method
                                           # [Ha]
                                           # q-vector fine grid for DOS calculation

dossmear 2e-5
ng2qpt 30 30 30
q2shft 0 0 0
```

Analogous to k-path definition

Phonons with DFPT in ABINIT (8/8)

- A `_PHBANDS.agr` file, containing the band structure information, that can be readily plotted with `xmgrace`, and a `*_PHDOS` file, containing the DOS information.
- For example, one gets for aluminum:



Electron-phonon coupling

Electron-phonon superconductivity

- Superconductor critical temperature:

$$T_c = \frac{\omega_{\log}}{1.2} \exp \left\{ - \frac{1.04(1 + \lambda)}{\lambda(1 - 0.62\mu^*) - \mu^*} \right\}$$

- μ^* : Morel-Anderson pseudo-potential, typically treated empirically (value 0.1-0.13)
- λ and ω_{\log} computed from Eliashberg function $\alpha^2 F(\omega)$
- Eliashberg function $\alpha^2 F(\omega)$ calculated from electron-phonon matrix elements $g_{n\mathbf{k}, \mathbf{k}+m\mathbf{q}}^\nu$, which describe the strength of the interaction between ions and electrons

e-ph. superconductivity in ABINIT (1/6)

- Approach very similar to phonon calculation (differences highlighted in red):
 1. Ground-state calculation (dataset 1) to get electron density $n(\mathbf{r})$
 - To be performed on full BZ and not only irreducible BZ ($kptopt=3$)
 - To get good e-ph matrix elements, you need fine k-grid (let's say 32x32x32) and tight tolerance ($tolwfr=1e-20$) (increase also max. number of SCF steps ($nstep=1000$))
 2. Calculation of linear response, dynamical matrices, and e-ph matrix elements on a grid of \mathbf{q} -vectors
 - ABINIT expects one and only one \mathbf{q} -vector per dataset
 - \mathbf{q} -vectors must be a subset of the k-vectors grid used in the GS calculation
 3. ABINIT will create a DDB (Derivative Data Base) file and a GKK file (e-ph matrix element database) per \mathbf{q} -vector that must be merged into a single DDB and GKK file with command line utility MRGDDB and MRGGKK
 4. Superconducting properties obtained by analyzing GKK data with ANADDB

e-ph. superconductivity in ABINIT (2/6)

- Steps 1-2: In ABINIT input file **response.abi**:
 - Add the following variables for phonon response datasets:

```
# e-ph related default settings
prepgkk 1                # do not reduce perturbations by symmetry
use_nonscf_gkk 0          # calculate all GKK matrix elements self-consistently
                           # not only the irreducible ones
prtgkk 1                  # print out e-ph matrix elements
prtwf 0                   # do not print out wave function (save disk space)
```

- Impose full BZ GS calculation, fine k-grid and tight wavefunction tolerance, and print out wavefunction for subsequent datasets:

```
kptopt 3  ← Must be same grid as perturbations k-grid  # do not account for symmetry when generating k-vectors
ngkpt 32 32 32 ← Fine grid needed for e-ph coupling  # number of grid points
                                                    # number of k vectors must be multiple of number of q vectors
tolwfr1 1.0d-20  # SCF convergence criterion: residual of the wavefunction
                                                    # a very tight value is needed to get good e-ph matrix elements
prtwf1 1          # print wavefunction for subsequent datasets
```

- Allows more self-consistent cycles by setting **nstep=1000**
- Everything else stays the same as for phonons!

e-ph. superconductivity in ABINIT (3/6)

- Steps 3: merge DDB files with `mrgddb < mrgddb.abi` and GKK files with `mrggkk < mrggkk.abi`. Example `mrggkk.abi` below (same “rules” as for `mrgddb` - remember, comments should NOT be included and are shown below just as explanation!).

```
3-mrggkk.abo          # Name of output file
0                     # output file format; 0: binary, 1: ascii (binary needed for ANADDB)
1-response/response_DS1_WFK # ground-state wave function file
0 435 435             # number of WF1 files (set to 0), of GKKs, and tot. num. of perts. (sum all GKKs)
1-response/response_DS2_GKK1 # name of 1WF and GKK files
1-response/response_DS2_GKK2
1-response/response_DS2_GKK3
1-response/response_DS3_GKK1
1-response/response_DS3_GKK2
1-response/response_DS3_GKK3
1-response/response_DS4_GKK1
1-response/response_DS4_GKK2
1-response/response_DS4_GKK3
```

Total number of perturbation.
In this case, each GKK file contains one q-vector. Hence, the number is the same as the one of GKK files.

- I have used a 16x16x16 q-grid → Computationally demanding!
- If we would merge two merged GKK files, we would need to specify the total number of q-vectors inside the GKK files.

e-ph. superconductivity in ABINIT (4/6)

- Step 4: Calculate superconducting properties with command `anaddb anaddb.abi`, with the ANADDB input file `anaddb.abi` written as below:

```
#####
# PREAMBLE
#####
output_file = "4-anaddb.abo"
ddb_filepath = "2-mrgddb.abo"
gkk_filepath = "3-mrggkk.abo"
eph_prefix = "./4-anaddb/anaddb"
ddk_filepath = "dummy"
outdata_prefix "./4-anaddb/anaddb"

#####
# FLAGS
#####
ifcflag 1                                # calculate interatomic force constants and
                                         # use them to interpolate phonon linewidths
                                         # and dynamical matrices at every q-vector

elphflag 1                              # analyze e-ph coupling

#####
# WAVE VECTOR (COARSE) GRID FROM CALCULATED DDB BASED ON WHICH IFCs CALCULATED
#####
brav 2                                # 1: all the lattices (FCC included, but results less accurate)
                                         # 2: FCC (help generation of special grid of q points)

ngqpt 16 16 16                        # Monkhorst-Pack grid to consider
                                         # (must correspond to DDB data or a subset of it)

nqshft 1                              # number of q-shifts
qlshft 0 0 0                          # shift vectors of coarse grid
```


e-ph. superconductivity in ABINIT (5/6)

```
#####  
# SUPERCONDUCTIVITY SETTINGS  
#####  
mustar 0.136  $\mu^*$ : Morel-Anderson pseudo-potential # Coulomb pseudopotential parameter  
  
#####  
# PATH TO INTERPOLATE PHONON LINEWIDTHS  
#####  
nqpath 7  
qpath 0.0 0.0 0.0  
      1/2 1/2 0.0  
      1   1   1  
      1/2 1/2 1/2  
      1/2 1/2 0.0  
      1/2 3/4 1/4  
      1/2 1/2 1/2
```

Must be specified because calculation
of phonon linewidths included in e-ph
analysis. Error raised otherwise.

e-ph. superconductivity in ABINIT (6/6)

- Eliashberg function $\alpha^2F(\omega)$ stored in `*_A2F` file



- Superconducting properties can be found in ANADDB output file `anaddb.abo`:

Superconductivity : isotropic evaluation of parameters from electron-phonon coupling.

```
mka2f: lambda <omega^2> =      4.709738E-07
mka2f: lambda <omega^3> =      5.358574E-10
mka2f: lambda <omega^4> =      6.386539E-13
mka2f: lambda <omega^5> =      7.861400E-16
mka2f: isotropic lambda =      4.700638E-01
mka2f: omegalog  =      8.686447E-04 (Ha)      2.742960E+02 (Kelvin)
mka2f: input mustar =      1.360000E-01
-mka2f: MacMillan Tc =      4.022671E-06 (Ha)      1.270258E+00 (Kelvin)
```

λ parameter

ω_{\log} parameter

T_c critical temperature

Remark

- Always check convergence of your phonon and e-ph related results with respect to variation of k- and q- grid size! You will get unphysical results (diverging T_c) if grids are too small.
- For e-ph coupling, you need large k- and q- grids because electron-phonon calculations presently implemented in ANADDB for superconductivity, determine isotropic quantities, which are averaged over the Fermi surface and summed over q-points.

Thank you for your attention!