

# First principles based interatomic potential generation for magnetic Iron



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# Outline

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# Why interatomic potential for iron?

- Iron originates in stars and is a supernova remnant.
- Iron causes earth's magnetic field.
- Iron alloys- elinvar, invar, various types of steels etc.
- Wide range applications of iron and its alloys.
  - Elinvar- balance springs for mechanical watches and chronometers.
  - Invar- motor valves, precision instruments, clocks, anti-magnetic watches, seismic creep gauges etc.
  - Iron and steel - construction industries, roads and railways, aerospace industry, ship building, heavy equipments like bulldozers, automobile industry, cooking utensils, surgical instruments and so on.
- Accuracy of molecular dynamics simulations.
- Inadequacy of reliable interatomic potentials.
  - Crack propagation analysis in  $\alpha$ -iron[1].
- Importance of magnetism.
  - Tight binding study performed by Guoqiang Liu et.al.[2].

# First principles method-DFT

- Hamiltonian operator for many-body time-independent Schrödinger equation [3].

$$\begin{aligned}\hat{H} &= -\frac{\hbar^2}{2m_e} \sum_i \nabla_i^2 - \frac{\hbar^2}{2M_I} \sum_I \nabla_I^2 - \sum_{i,I} \frac{Z_I e^2}{|r_i - R_I|} \\ &+ \frac{1}{2} \sum_{i \neq j} \frac{e^2}{|r_i - r_j|} + \frac{1}{2} \sum_{I \neq J} \frac{Z_I Z_J e^2}{|R_I - R_J|} \\ &= T_e + T_N + V_{Ne}(r, R) + V_{ee}(r) + V_{NN}(R)\end{aligned}$$

- Using Born-Oppenheimer approximation,  $T_N$  can be set to zero.
- Hence ,the many-body time-independent Schrödinger equation.

$$\hat{H} |\Psi_i(r)\rangle = E |\Psi_i(r)\rangle$$

- Solving above Schrödinger equation is not feasible.

# First principles method-DFT

- Hence to solve a non-relativistic many-body system, density functional theory (DFT) is used.
- The works of Pierre Hohenberg , Walter Kohn and Lu Jeu Sham lead to DFT.
- DFT uses electron densities which are defined using Kohn - Sham wavefunctions.
- To calculate the ground state energy of a system, an approximate  $E_{xc}$  functional determining exchange-correlation energy of electrons, is used (GGA, LDA, Hybrid functionals).

## Potfit package

- Potfit uses force-matching algorithm to generate inter atomic potentials with first principles data as reference [4].
- Designed primarily to use VASP output as reference data.
- Potfit requires three input files.
  - Initial potential file.
  - Parameter file.
  - Configuration file.
- Number of analytic functions to be used in initial potential file for an ADP potential generation is  $N(3N+7)/2$ .
- Here for a chosen interaction model, potential generation involves two major tasks.
  - Use force-matching algorithm to calculate potential at every parameter set.
  - Parameterize the potential model by optimizing/minimizing the target function thus providing optimized parameter set.

## Potfit package

- Target function is optimized using different optimization algorithms.
- Optimization algorithms used are
  - Simulated annealing.
  - Differential evolution.
  - Powell's least square method(conjugate gradient).
- Target function[\[4\]](#).

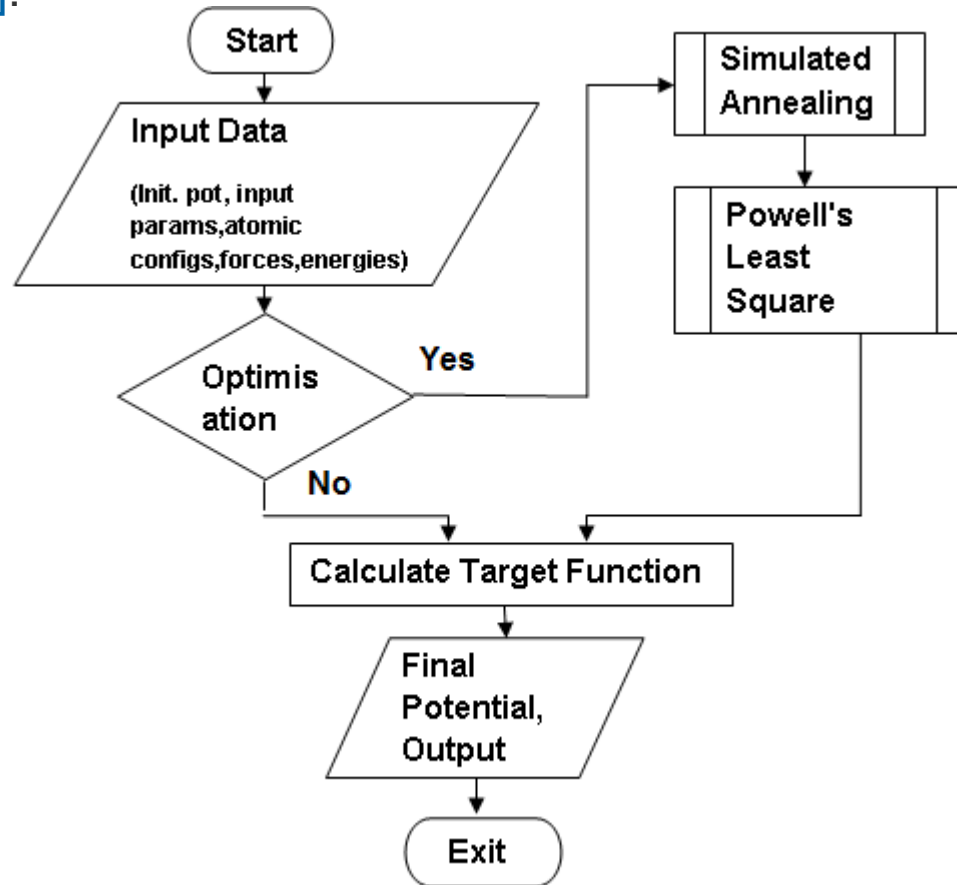
$$Z(\xi) = Z_F(\xi) + Z_C(\xi),$$

where

$$Z_F(\xi) = \sum_{k=1}^m u_k (F_k(\xi) - F_k^0)^2,$$
$$Z_C(\xi) = \sum_{r=1}^{N_c} w_r (A_r(\xi) - A_r^0)^2.$$

# Potfit package

- Potfit Flowchart [4].





# Angular Dependent Potential

- Formalism.

$$E_{\text{tot}} = \frac{1}{2} \sum_{i,j;i \neq j}^N \Phi_{ij}(r_{ij}) + \sum_i F_i(n_i) + \frac{1}{2} \sum_{i,\alpha} (\mu_i^\alpha)^2 + \frac{1}{2} \sum_{i,\alpha,\beta} (\lambda_i^{\alpha\beta})^2 - \frac{1}{6} \sum_i \nu_i^2$$

- The first  $\Phi_{ij}(r_{ij})$  represents pair interaction.
- The second term  $F_i(n_i)$  represents embedding function.
- The third  $\mu_i^\alpha$ , fourth  $\lambda_i^{\alpha\beta}$  and fifth  $\nu_i$  terms represent vectors, tensors and trace of  $\lambda_i^{\alpha\beta}$ . respectively.

## DFT data set

- Elk FP-LAPW code is used [5].
- Allotropes of iron under consideration:  $\alpha$ -iron – FM, AFM, NM,  $\gamma$ -iron – AFM, NM,  $\epsilon$ -iron – FM, AFM, NM.
- $E_{XC}$  functional GGA-PBE [6] is used.
- $E(V)$  curves are obtained just around the equilibrium volume (quadratic region) using spinpolarized calculations.
- The configurations resulting in  $E(V)$  curves form the reference data set for potfit.
- Verify the nature of  $E(V)$  and  $E(M)$  curves.
- Calculate cohesive energies of all configurations.
- Cohesive energy is calculated as

$$E_{coh} = \text{Isolated Fe atom energy} - \text{Fe atom energy in crystal.}$$

## Elk output conversion

- Potfit is compatible with VASP output format.
- Hence, elk output should be converted to potfit compatible format.
- Units of force, cohesive energy, geometry information in potfit input files are eV/Å, eV, Å respectively.

## Potfit potential generation

- To generate ADP potential, represent the five interaction terms of ADP formalism using analytic functions 'eopp' (empirical oscillating potentials) [7], 'csw2' [8], 'bjs' [9].

'eopp' with six parameters

$$V(r) = \frac{C_1}{r^{\eta_1}} + \frac{C_2}{r^{\eta_2}} \cos(kr + \varphi)$$

'csw2' with four parameters

$$\rho(r) = \frac{1 + a_1 \cos(\alpha r + \varphi)}{r^\beta}$$

'bjs' with three parameters

$$F(n) = F_0[1 - \gamma \ln n]n^\gamma + F_1 n$$

## Potfit potential generation

- Initialize the parameter set i.e initial potential file.
- Configuration and parameter files are prepared.
- Configuration file contains reference data from DFT and parameter file contains input parameters like optimization methods, energy and stress weights, output potential format, number of steps and so on.
- Here, simulated annealing and Powell's least square method are used.
- Force weight of '1' and energy weight of '100' are used in this work.
- Vary the parameter for annealing temperature, parameter set and reference data set (if needed) until satisfactory potential is obtained.
- Test the potential through molecular dynamics calculations.

## Testing the quality of potfit potential

- LAMMPS [10] package is used to perform MD calculations.
- MD calculations performed.
  - Cohesive energy and lattice parameter.
  - $E(V)$  curves.
  - Elastic constants.
  - Relaxed monovacancy formation energy.
  - Surface energies ( $E_{(100)}$ ,  $E_{(110)}$ ,  $E_{(111)}$  in  $\alpha$  and  $\gamma$  iron).
  - Generalized stacking fault energy curve (GSFE curve for  $\alpha$ -iron).

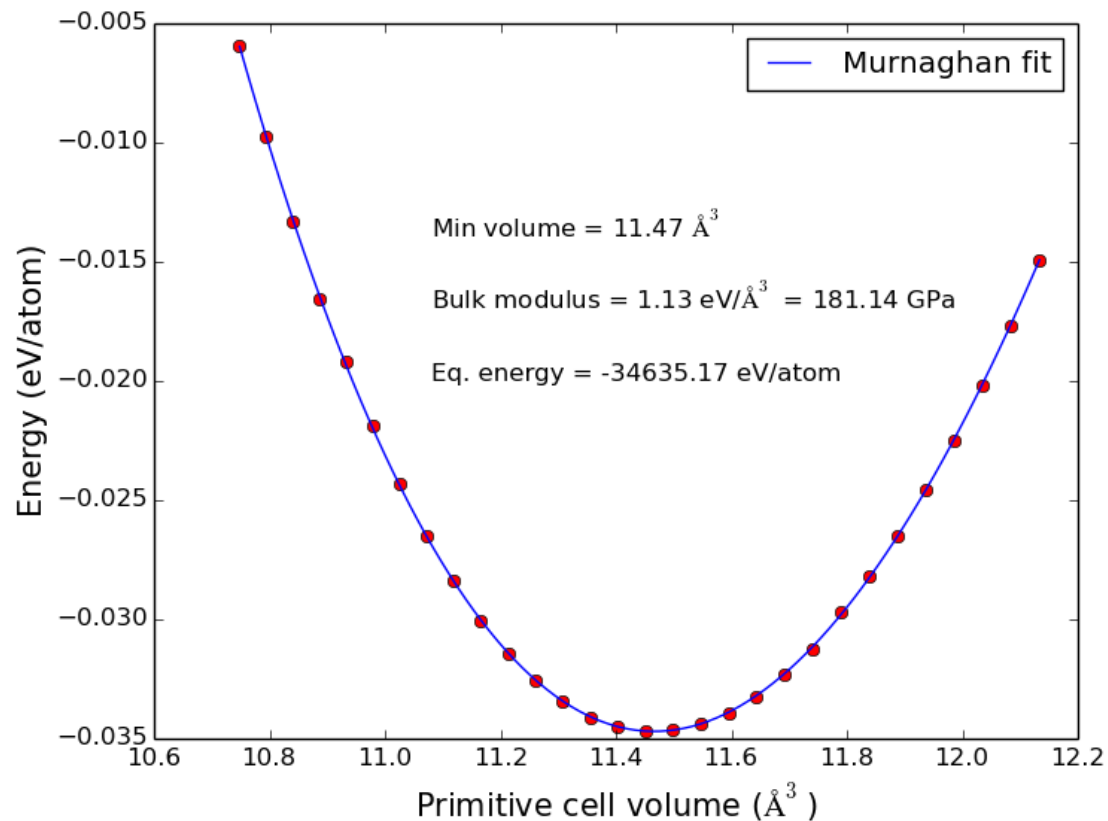
## Results-DFT data set

- Equilibrium properties of relaxed structures.

Strcuture	$V_0$	$E_{0,coh}$	$B_0$	Ref. $V_0$	Ref. $E_{0,coh}$	Ref. $B_0$
-	$\text{\AA}^3$	eV/atom	Gpa	$\text{\AA}^3$	eV/atom	Gpa
bcc-NM	10.478	-3.49	269.94	10.35[11]	-	288[11]
bcc-FM	11.465	-4.11	181.14	11.36[12]	-4.28[12]	189[12]
bcc-AFM	11.514	-3.96	190.98	10.97[13]	-	176[13]
fcc-NM	10.17	-3.90	278.94	10.33[11]	-	283[11]
fcc-AFM	11.26	-4.11	142.29	11.30[12]	-	133[12]
hcp-NM	10.12	-3.96	293.31	10.23[11]	-	292[11]
hcp-FM	10.12	-3.94	292.59	10.55[11]	-	241[11]
hcp-AFM	10.383	-3.97	233.72	10.43[14]	-	213[14]

## Results-DFT data set

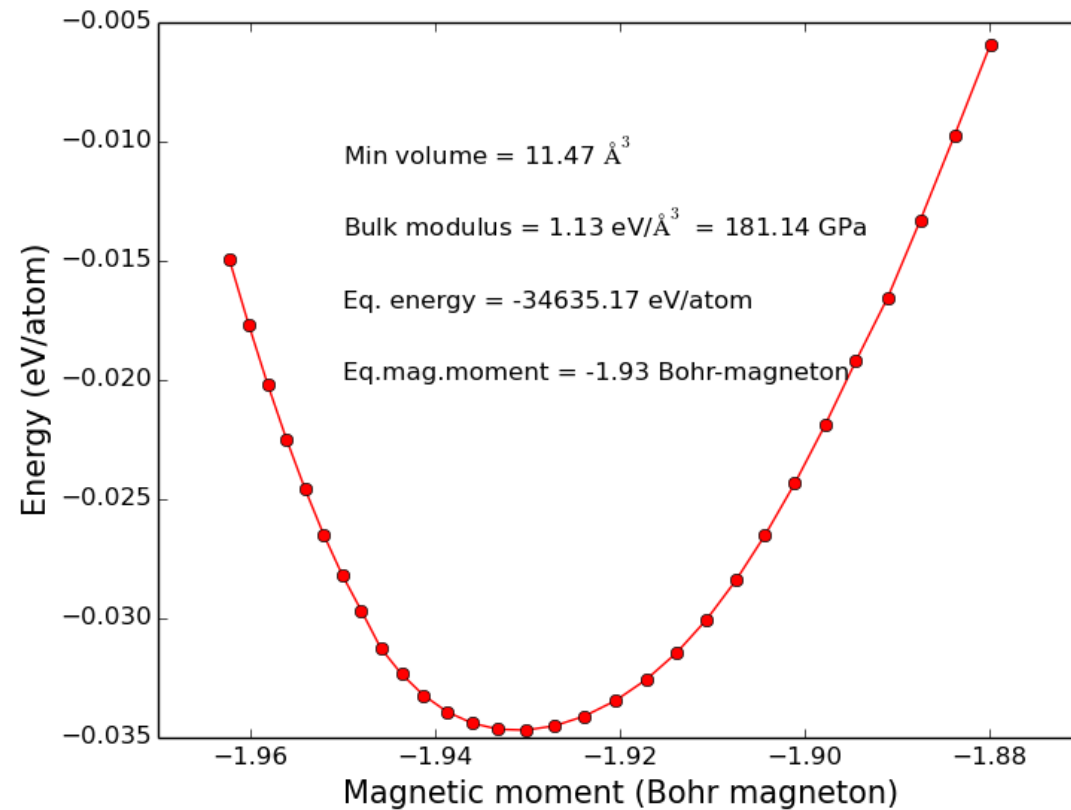
- The FM  $\alpha$ -iron study.
- $E(V)$  curve.





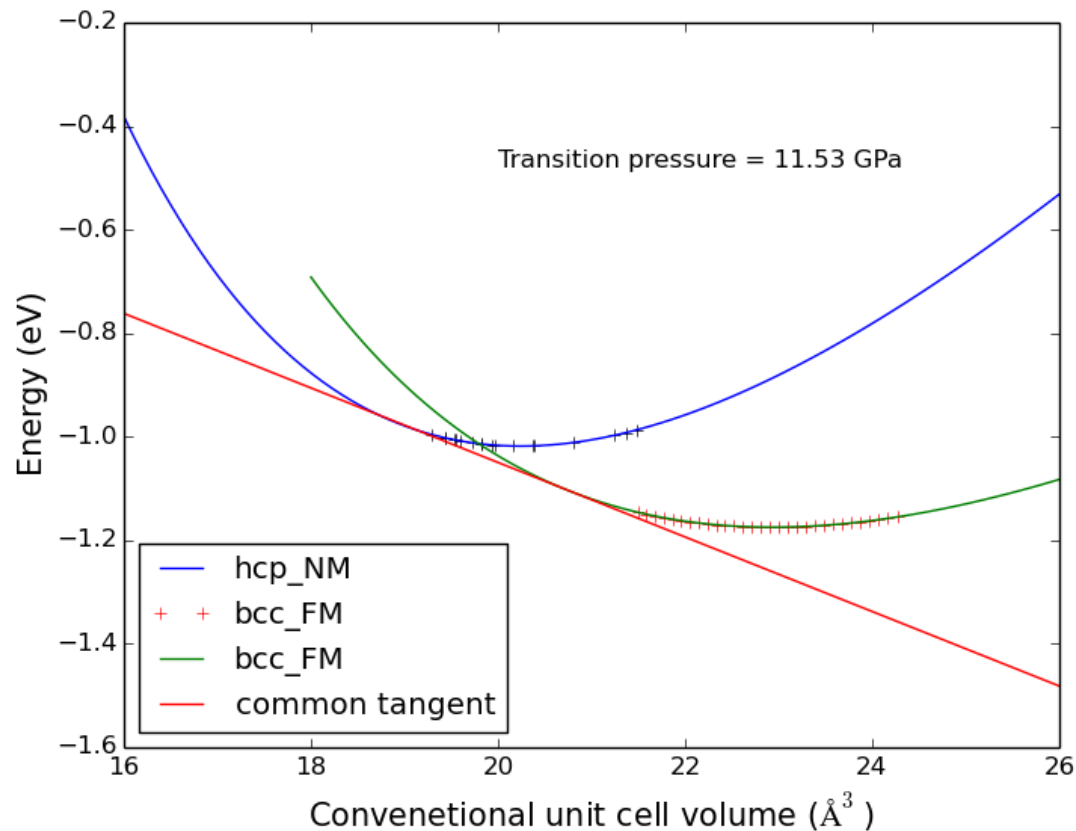
## Results-DFT data set

- The FM  $\alpha$ -iron study.
- $E(M)$  curve.



## Results-DFT data set

- The FM  $\alpha$ -iron to NM  $\epsilon$ -iron phase transition study[15].



## Results-Potfit potential for $\alpha$ -iron.

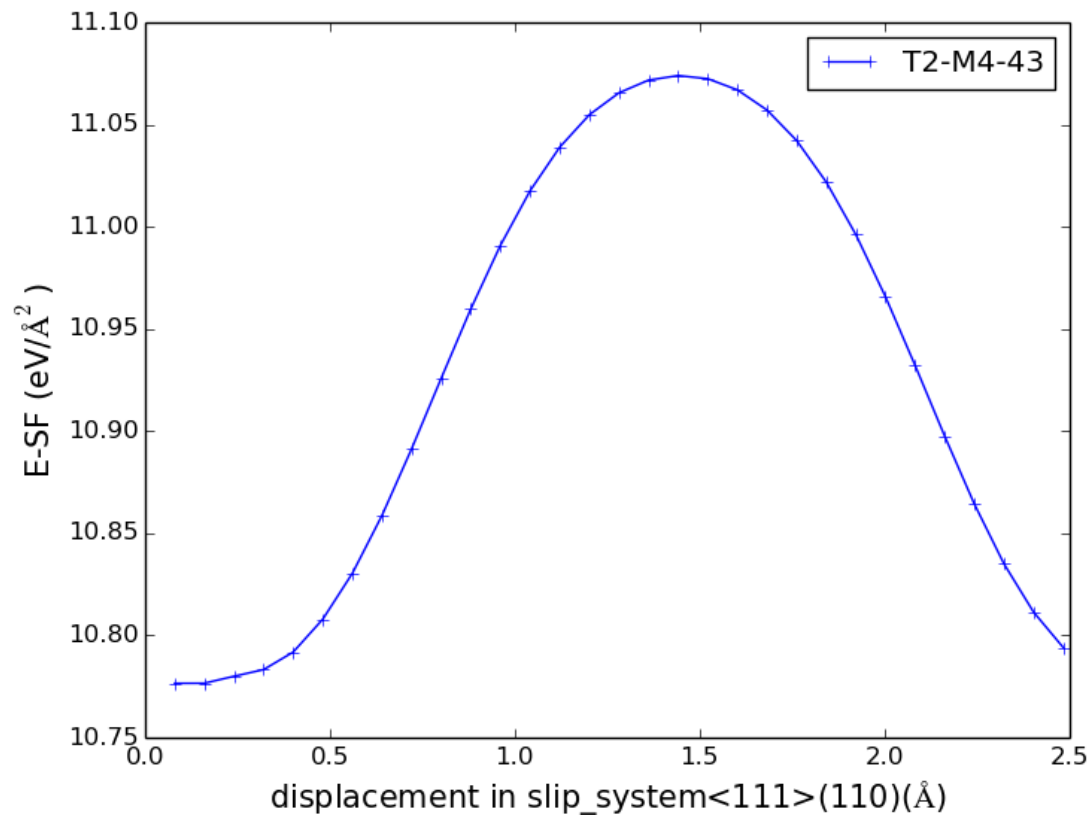
- Some potentials reproduce elastic constants very accurately but fail to reproduce surface energies.
- Some other potentials reproduce elastic constants poorly and reproduce the ordering of surface energies precisely but not the magnitudes of surface energies.
- So of all the  $\alpha$ -iron potentials generated in this work, potential „T2-M4-43“ behaves pretty well in all the cases.
- Comparison of properties calculated with potential „T2-M4-43“ and reference results is shown in following table.

## Results-Potfit potential for $\alpha$ -iron.

Properties	Units	T2-M4-43 pot	Reference
$E_{coh}$	eV/atom	-4.07	-4.28[16]
$V_0$	$\text{\AA}^3$	11.42	11.78 <sup>Expt</sup> [16]
$C_{11}$	Gpa	248.71	242 <sup>Expt</sup> [16]
$C_{12}$	Gpa	161.15	146.5 <sup>Expt</sup> [16]
$C_{44}$	Gpa	119.53	112 <sup>Expt</sup> [16]
$B_0$	Gpa	190.34	189 <sup>DFT</sup> [12]
$SM-I$	Gpa	119.53	-
$SM-II$	Gpa	43.78	-
$\mu$	Gpa	0.393	-
$E_{vac}$	eV/atom	1.22	1.53[16]
$E_{(100)}$	J/m <sup>2</sup>	3.30	2.463 <sup>MD</sup> [16]
$E_{(110)}$	J/m <sup>2</sup>	3.213	2.358 <sup>MD</sup> [16]
$E_{(111)}$	J/m <sup>2</sup>	3.56	2.658 <sup>MD</sup> [16]

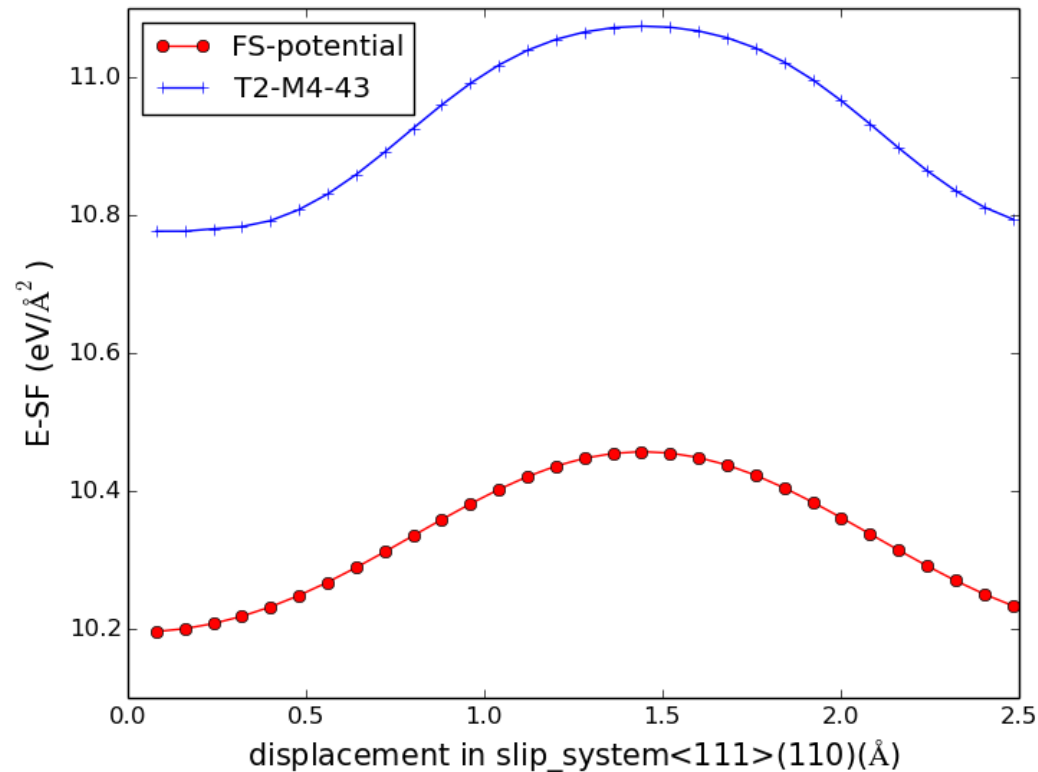
## Results-Potfit potential for $\alpha$ -iron.

- Generalized stacking fault energy curve.



## Results-Potfit potential for $\alpha$ -iron.

- Comparison of GSFE using potfit potential T2-M4-43 and Finnis Sinclair potential [18].



## Results-Potfit potential for $\gamma$ -iron.

- Just as in the case of  $\alpha$ -iron potentials, of all the  $\gamma$ -iron potentials generated in this work, potential „T22-M6-2“ behaves pretty well in all the cases and is comparable to Analytic Bond Order Potential(ABOP) developed by M Müller et.al [\[12\]](#).
- Comparison of properties calculated with potential „T22-M6-2“ and ABOP, reference results is shown in following table.

## Results-Potfit potential for $\gamma$ -iron.

Properties	Units	T22-M6-2 pot	Reference
$E_{coh}$	eV/atom	-4.11	-
$V_0$	$\text{\AA}^3$	11.25	11.3 <sup>ABOP</sup> [12]
$C_{11}$	Gpa	217.5	204 <sup>ABOP</sup> [12]
$C_{12}$	Gpa	149.096	144 <sup>ABOP</sup> [12]
$C_{44}$	Gpa	115.125	101 <sup>ABOP</sup> [12]
$B_0$	Gpa	171.89	164 <sup>DFT</sup> [12]
$SM-I$	Gpa	115.125	-
$SM-II$	Gpa	34.2	-
$\mu$	Gpa	0.406	-
$E_{vac}$	eV/atom	1.128	1.7 <sup>Expt</sup> [16]
$E_{(100)}$	J/m <sup>2</sup>	1.678	2.463 <sup>MD</sup> [16]
$E_{(110)}$	J/m <sup>2</sup>	2.429	2.358 <sup>MD</sup> [16]
$E_{(111)}$	J/m <sup>2</sup>	1.397	2.658 <sup>MD</sup> [16]



## Results-Potfit potential for $\epsilon$ -iron.

- Potfit generated –iron potential „T30-M5-1“ is tested only for elastic constants,  $E(V)$  curve.

Properties	Units	T30-M5-1 pot	Reference
$E_{coh}$	eV/atom	-4.09	-
$V_0$	$\text{\AA}^3$	21.5	20.46 <sup>DFT</sup> [17]
$C_{11}$	Gpa	501.94	556 <sup>DFT</sup> [17]
$C_{12}$	Gpa	364.83	300 <sup>Expt</sup> [17]
$C_{44}$	Gpa	61.54	248 <sup>DFT</sup> [17]
$C_{33}$	Gpa	662.51	647 <sup>DFT</sup> [17]
$C_{13}$	Gpa	162.77	143 <sup>DFT</sup> [17]
$C_{66}$	Gpa	68.47	
$B_0$	Gpa	263.16	297 <sup>DFT</sup> [17]

## Results-MD surface energy calculation.

- Three different surface energy calculation methods have been employed.
  - Method of calculation using sufficient layers along direction of surface.
  - Periodic slab geometry method.
  - Intercept method.
  
- All these methods are congruent i.e yield similar results for surface energies.

## Conclusion

- Some reference data which cannot be obtained from experimental procedures can be calculated using DFT with better accuracy.
- Hence, large amount of information can be used as reference data for potfit making it possible to generate more realistic interatomic potential for MD calculations.
- Application of apt zeeman field during spinpolarized calculation is paramount for obtaining correcting magnetic phase with correct equilibrium magnetic moment.
- FM  $\gamma$ -iron spinpolarized and FSM calculations indicate the existence of non-collinear magnetism.
- GGA-PBE functional is not non-collinear unlike LSDA and hence it poses convergence problems in non-collinear spinpolarized calculations.
- Importance of 3d-electrons which are shielded by only two 4s electrons.

# Conclusion

- Quality of potfit generated potential depends on quality of reference data from DFT.
- Optimization algorithms like simulated annealing help to obtain a better global minimum thereby facilitating generation of more realistic interatomic potentials.
- Introduction of a self-consistent electron density term in potential model might be the key for generating a common potential for different phases of iron.
- Transferability of potential depends on the configurations in reference data set.

## Future work

- If different defect energies are calculated in DFT and used as part of reference data in potfit, a very good potential can be obtained. Then it is possible to use this potential to understand crack propagation behaviour.
- It would also be easier to optimize the parameters corresponding to non-central bonding terms in ADP formalism if defect energies are included in reference data.

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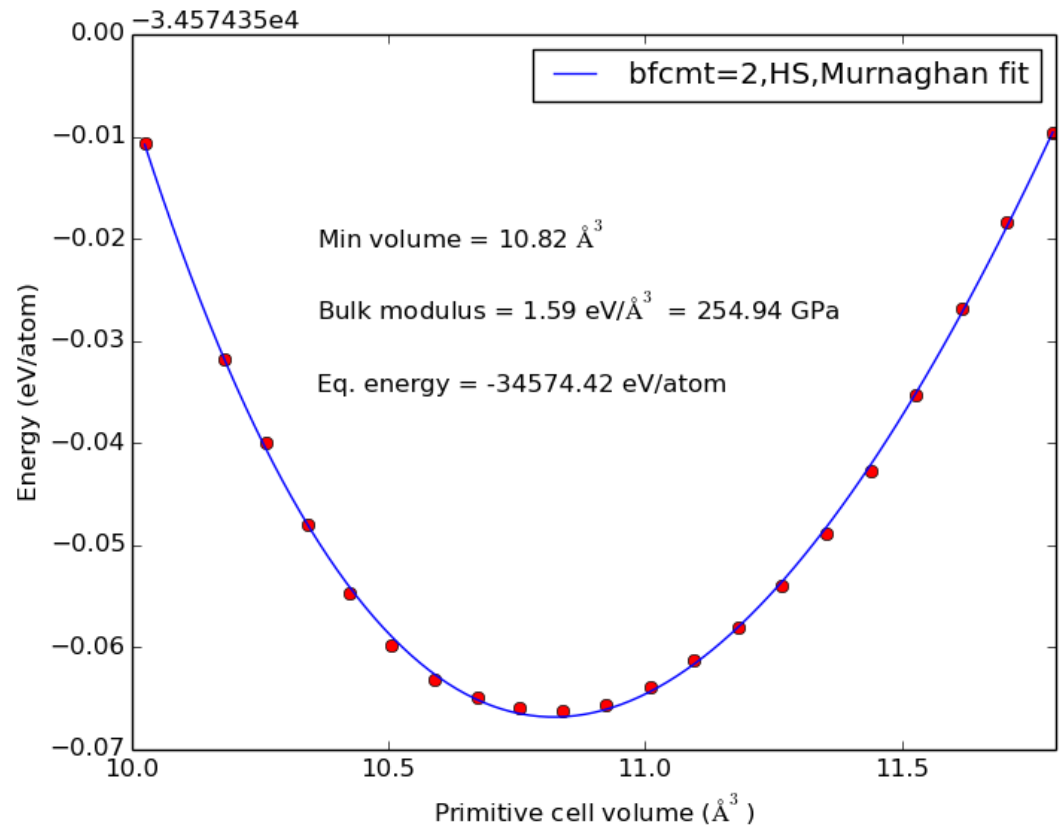
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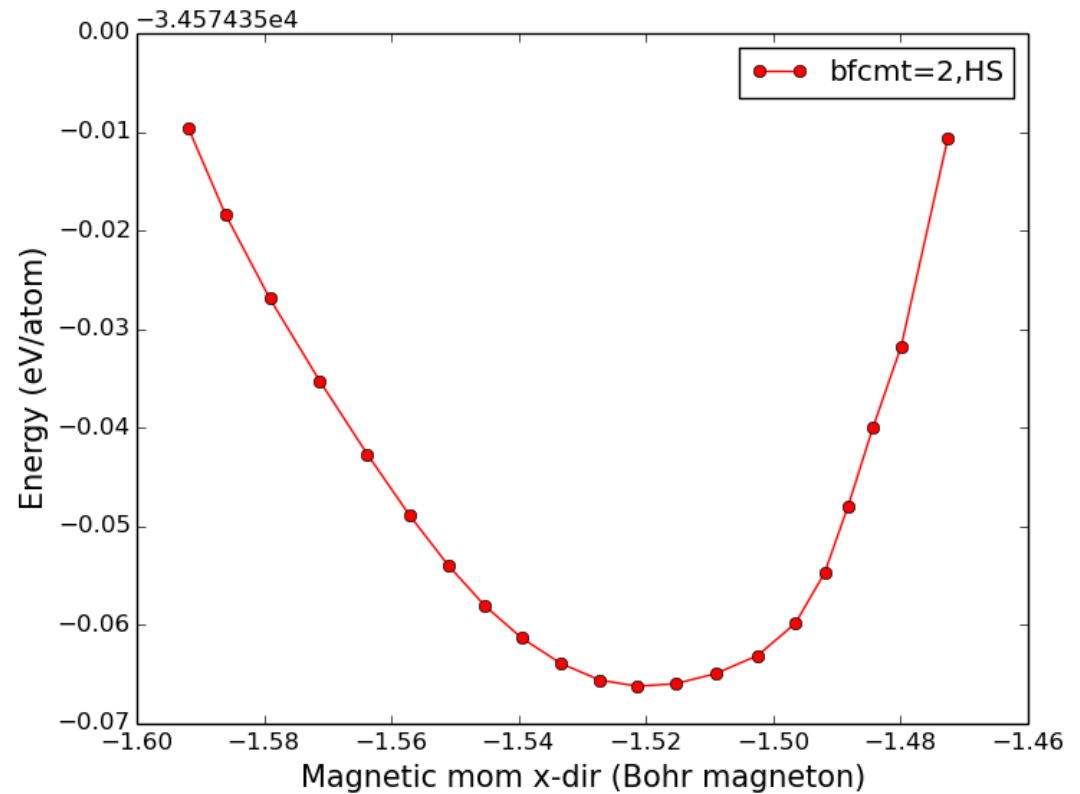
## Appendix

- FM  $\gamma$ -iron is studied using LSDA  $E_{XC}$  functional.
- HS-phase  $E(V)$  curve



## Appendix

- FM  $\gamma$ -iron is studied using LSDA  $E_{XC}$  functional.
- HS-phase  $E(M)$  curve



## Appendix

- FM  $\gamma$ -iron is study.
- Here, equilibrium volume of  $10.89 \text{ \AA}^3$  (primitive cell), bulk modulus of 254.94 Gpa and magnetic moment of  $2.64 \mu_B$  are observed.
- These are in agreement with other first principles results [\[19\]](#).



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