

parishcm@ornl.gov



# A brief discussion on X-ray spectrum image analysis

**Chad M. Parish**

**Radiation Effects and Microstructural Analysis Group (REMAG)**

**Materials Science and Technology Division**

**Oak Ridge National Laboratory**

ORNL is managed by UT-Battelle, LLC for the US Department of Energy

**Research sponsored the U.S. Department of Energy,  
Office of Fusion Energy Sciences under contact DE-  
AC05-00OR22725 with UT-Battelle LLC.**



# Bibliography

- Burke, M. G., M. Watanabe, D. B. Williams and J. M. Hyde (2006). "Quantitative characterization of nanoprecipitates in irradiated low-alloy steels: advances in the application of FEG-STEM quantitative microanalysis to real materials." *Journal of Materials Science* 41(14): 4512-4522.
- Keenan, M. R. (2007). Multivariate analysis of spectral images composed of count data. *Techniques and Applications of Hyperspectral Image Analysis*. H. F. Grahn and P. Geladi. Chichester, John Wiley & Sons: 89-126.
- Keenan, M. R. (2009). "Exploiting Spatial-Domain Simplicity in Spectral Image Analysis." *Surface and Interface Analysis* 41: 79-87.
- Keenan, M. R. and P. G. Kotula (2004). "Accounting for Poisson noise in the multivariate analysis of ToF-SIMS spectrum images." *Surface and Interface Analysis* 36(3): 203-212.
- Kotula, P. and M. Van Benthem (2015). "Revisiting noise scaling for multivariate statistical analysis." *Microscopy and Microanalysis* 21(S3): 1423-1424.
- Kotula, P. G. and M. R. Keenan (2002). "Spectral imaging: towards quantitative X-ray microanalysis." *Microscopy and Microanalysis* 8: 440-441.
- Kotula, P. G. and M. R. Keenan (2006). "Application of multivariate statistical analysis to STEM X-ray spectral images: Interfacial analysis in microelectronics." *Microscopy and Microanalysis* 12(6): 538-544.
- Kotula, P. G., M. R. Keenan and J. R. Michael (2003). "Automated analysis of SEM X-ray spectral images: a powerful new microanalysis tool." *Microscopy and Microanalysis* 9(1): 1-17.
- Kotula, P. G., D. O. Klenov and H. S. von Harrach (2012). "Challenges to Quantitative Multivariate Statistical Analysis of Atomic-Resolution X-Ray Spectral." *Microscopy and Microanalysis* 18(4): 691-698.
- Parish, C. M. (2011). Multivariate Statistics Applications in Scanning Transmission Electron Microscopy X-Ray Spectrum Imaging. *Advances in Imaging and Electron Physics*, Vol 168. P. W. Hawkes. 168: 249-295.
- Parish, C. M., K. G. Field, A. Certain and J. P. Wharry (2015). "Application of STEM characterization for investigating radiation effects in BCC Fe-based alloys." *Journal of Materials Research* 30(9): 1275-1289.
- Smentkowski, V. S., S. G. Ostrowski, E. Braunstein, M. R. Keenan, J. A. T. Ohlhausen and P. G. Kotula (2007). "multivariate statistical analysis of three-spatial-dimension TOF-SIMS raw data sets." *Analytical Chemistry* 79(20): 7719-7726.
- Smentkowski, V. S., S. G. Ostrowski and M. R. Keenan (2009). "A comparison of multivariate statistical analysis protocols for ToF-SIMS spectral images." *Surface and Interface Analysis* 41: 88-96.
- Stork, C. L. and M. R. Keenan (2010). "Advantages of clustering in the phase classification of hyperspectral materials images." *Microscopy and Microanalysis* 16: 810-820.

# X-rays in electron microscopy: what, why, how

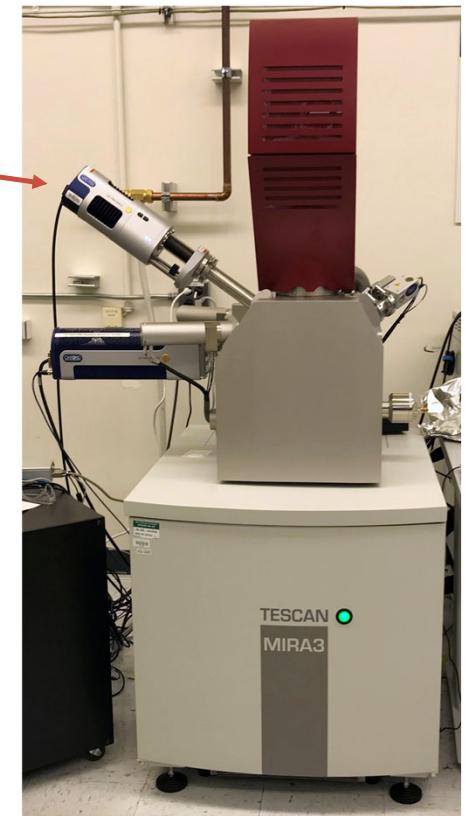
- Electron microscopy always generates X-rays—might as well put them to work!

Typical STEM  
(Four SDD-EDS, internal)



SDD-EDS: Silicon drift detector  
energy dispersive X-ray  
spectrometer

Typical SEM



Equivalent acronyms (interchangeable):

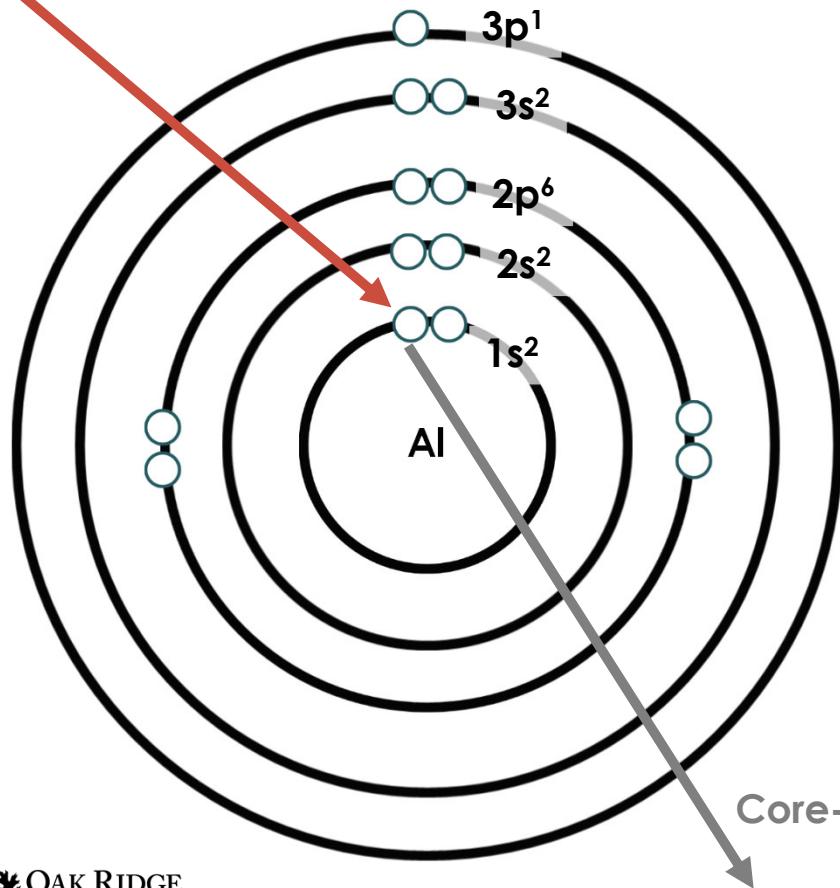
★ EDS  
EDX  
EDXS  
XEDS

Trade name, not acceptable:  
EDAX

(I have seen *all* of these in the literature)

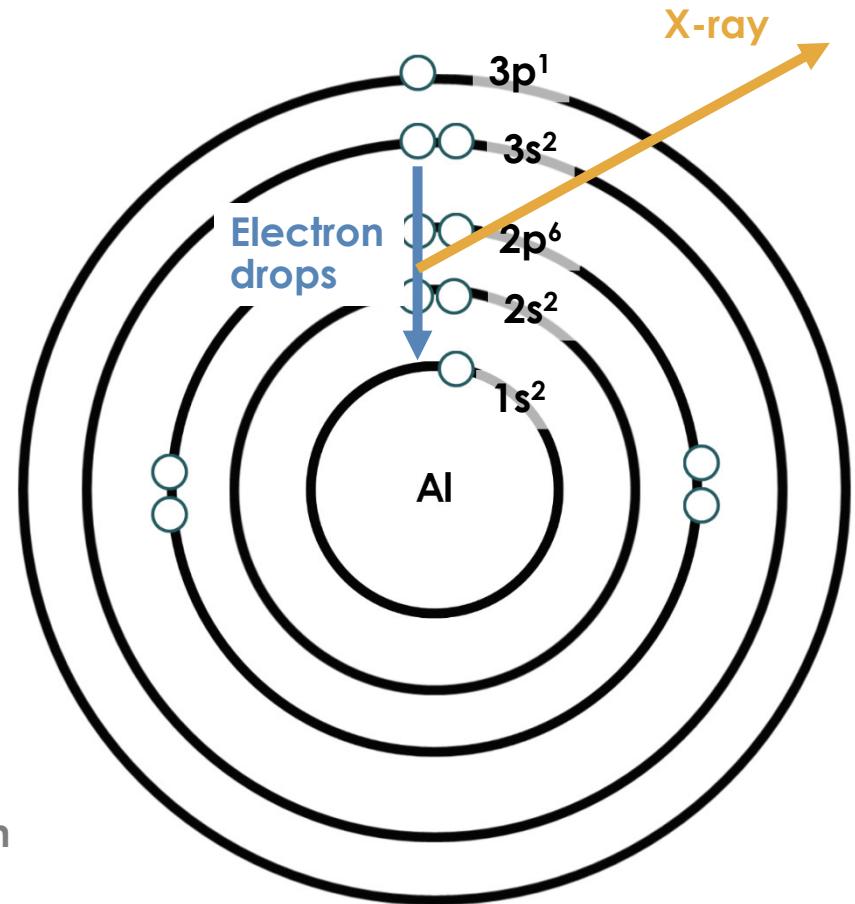
# X-rays in electron microscopy: what, why, how

High-energy electron



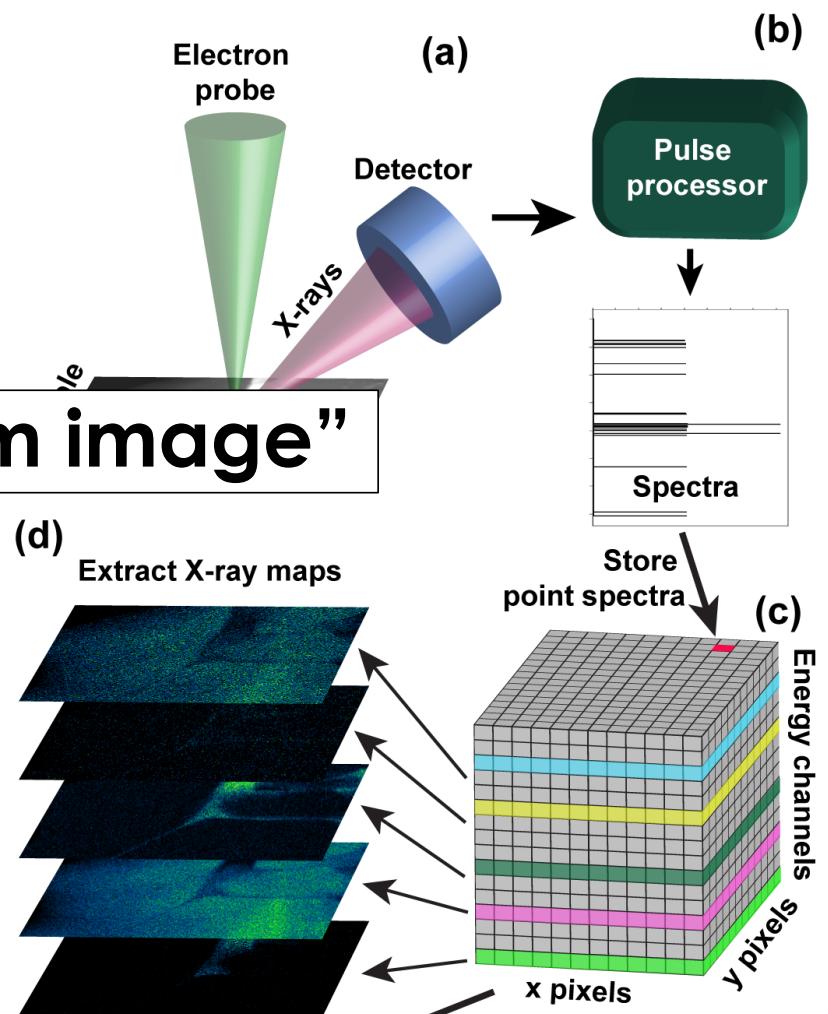
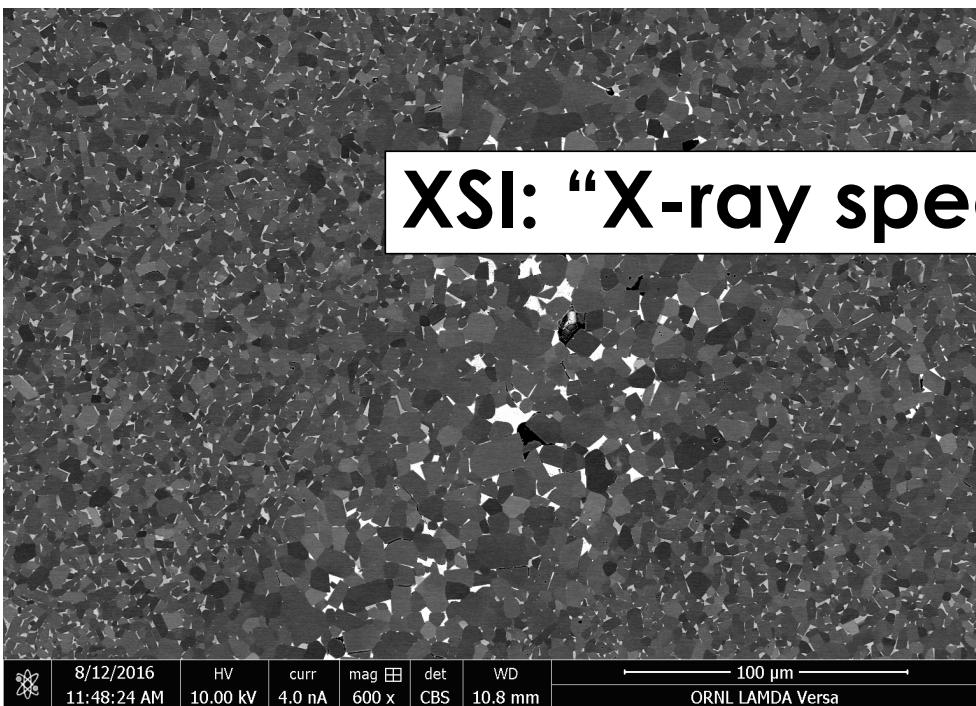
Electron drops

X-ray



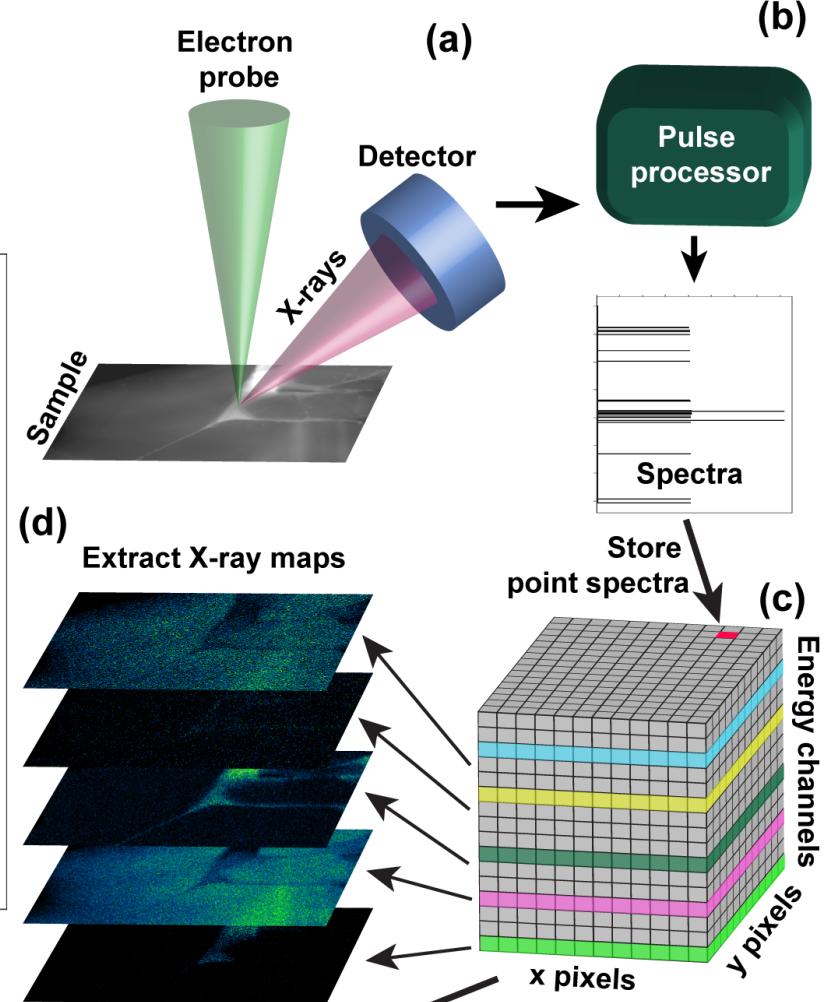
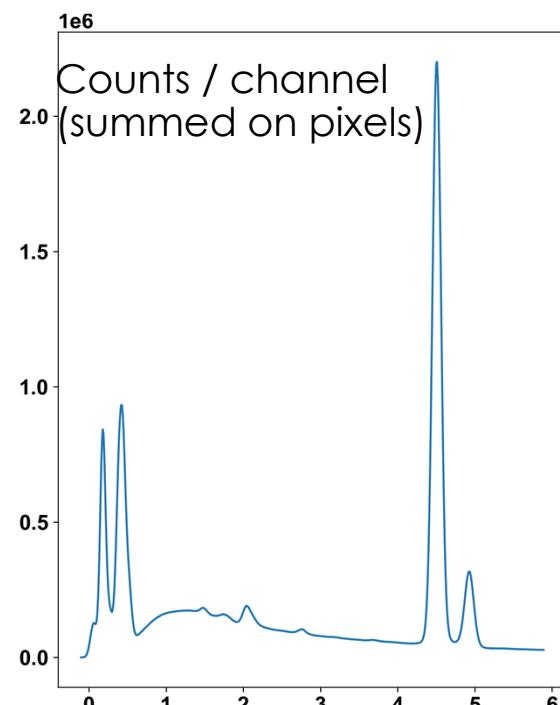
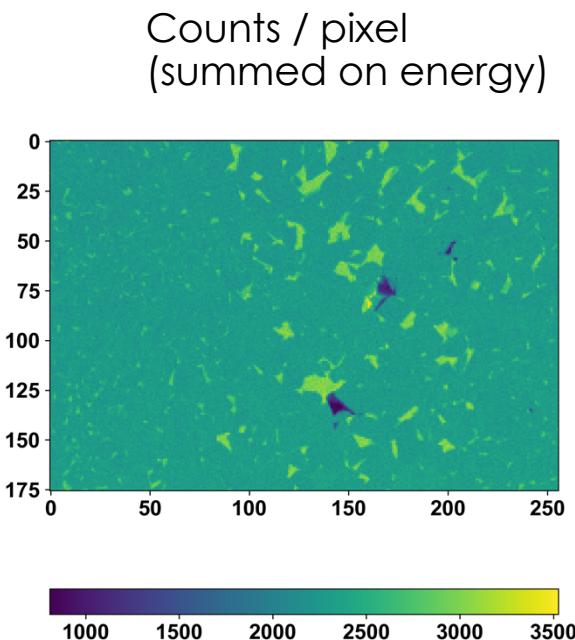
# X-rays in electron microscopy: what, why, how

- Scan a beam across a surface, collect the spatially and energy-resolved X-ray



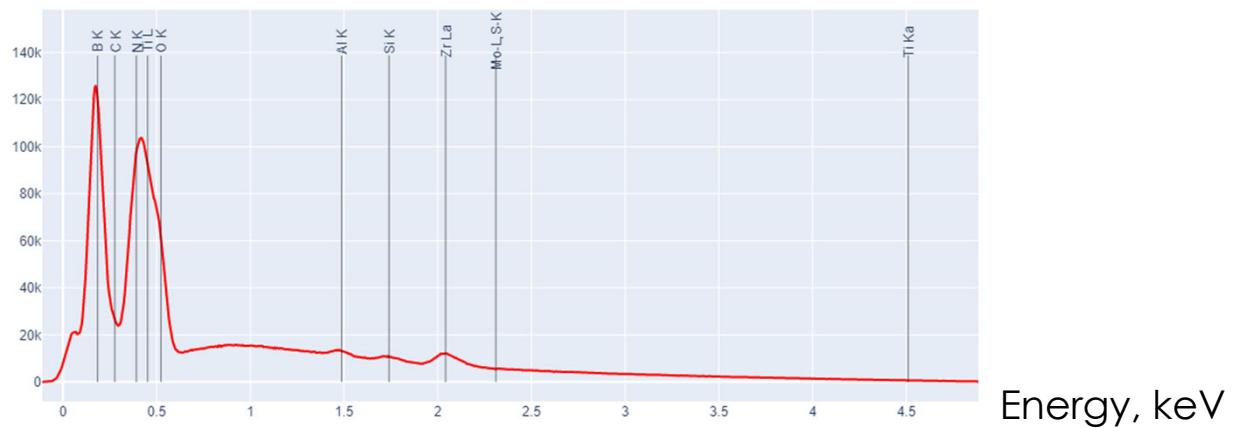
# X-rays in electron microscopy: what, why, how

- Scan a beam across a surface, collect the spatially and energy-resolved X-ray



# X-rays in electron microscopy: what, why, how

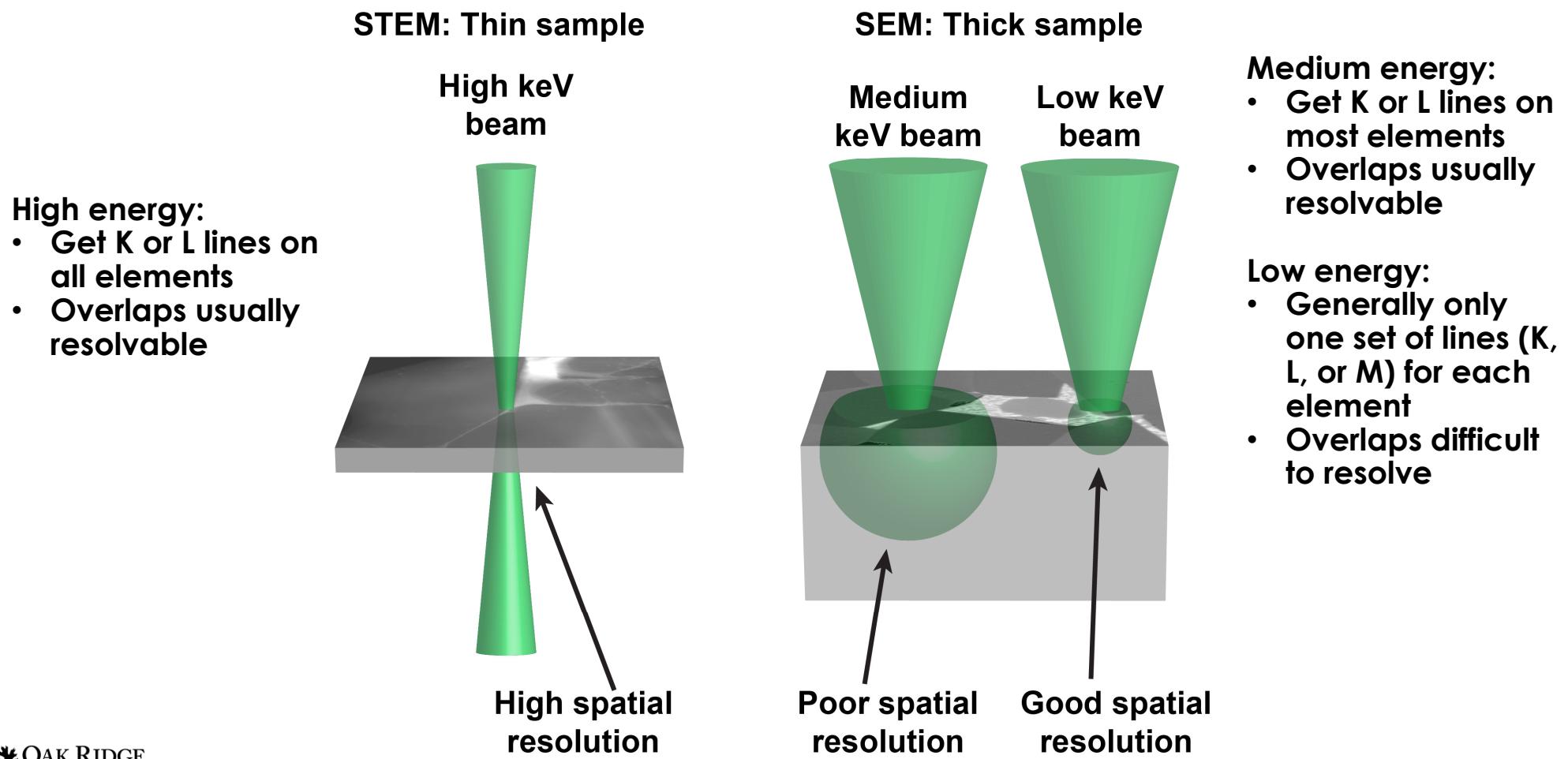
- ***Characteristic and background (Bremsstrahlung) X-rays are always present***



- **When X-rays are good choices:**
  - Elements  $\geq$ C or so, especially elements  $\geq$ Al or so
  - Amounts  $\sim$ 1 wt% or more
- **When X-rays are bad choices:**
  - Be, Li (modern detector); <Al (old detectors)
  - Low (<<1 wt%) amounts

# X-rays in electron microscopy: what, why, how

## Spatial resolution: three broad cases



# Applying ML to X-ray spectrum image data

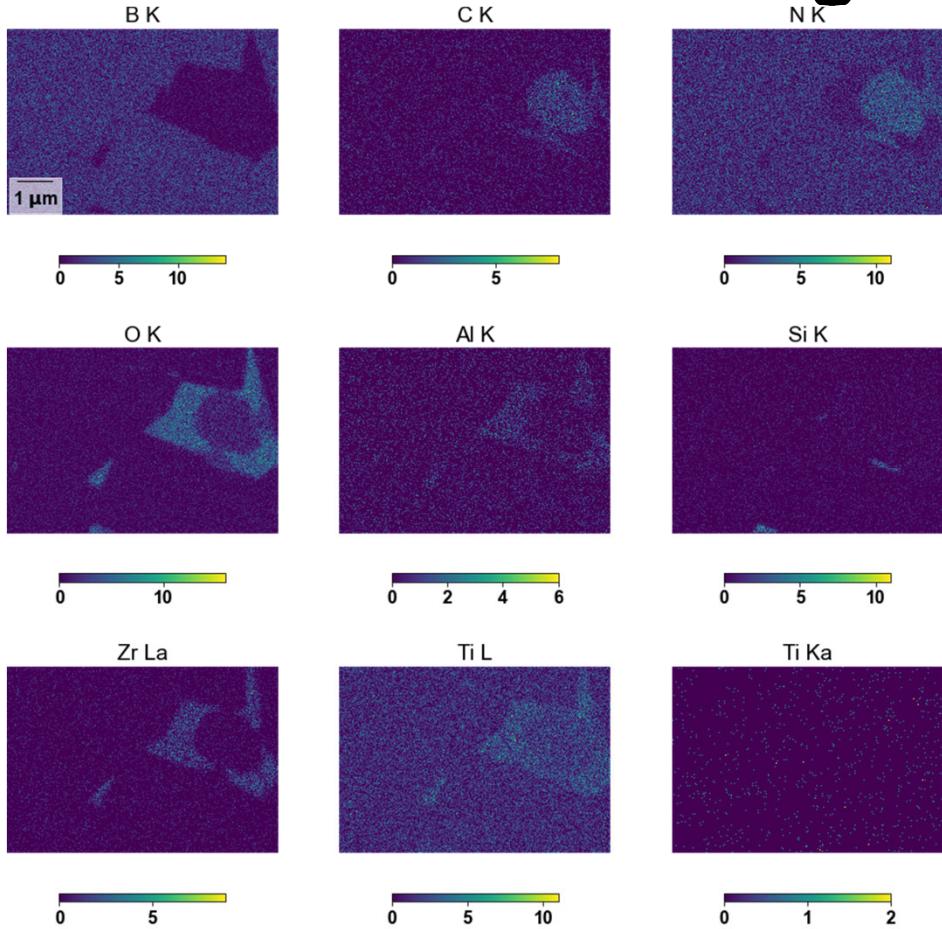
- Machine learning: PCA, SVD, NMF, K-means, DLNN, SVM, and umpteen bazillion others...
- Unsupervised methods generally used:
  - Forward models exist, but are not high enough fidelity to populate databases
  - We generally want to know *a posteriori* what was in our sample
  - X-ray physics is well-approximated by linear combinations of compound/element spectra, so bilinear factor models match the physics wonderfully!

$$D \approx AS^T$$

Data  $\approx$  (abundance maps) @ (endmember spectra transpose)

→ *Singular value decomposition-based methods are preferred*

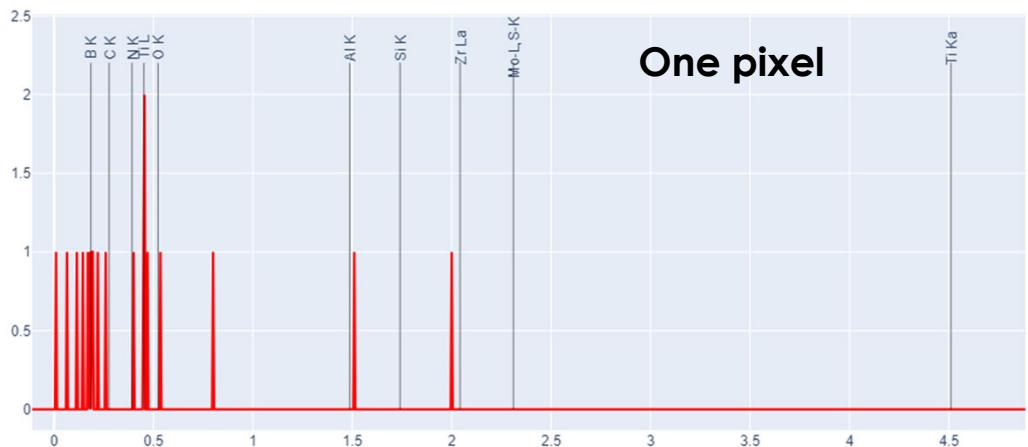
# Real XSI data challenges: sparse, large, noisy



$\text{TiB}_2$  sintered ceramic, 5 kV



Sum spectrum



One pixel

# Workflow

- We'll go through this step by step

$$I(x, y|E) \rightarrow D_{n \times m}$$

$$\hat{D} = GDH$$

$$\hat{D} \approx U\Sigma V^T$$

$$\hat{D} \approx U\Sigma V^T = \hat{A}\hat{S}^T$$

$$\mathbf{S} = \mathbf{H}^{-1}\hat{\mathbf{S}}$$

$$\mathbf{A} = \mathbf{G}^{-1}\hat{\mathbf{A}}$$

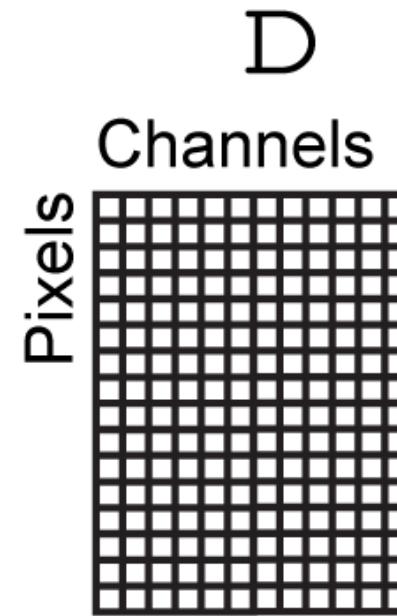
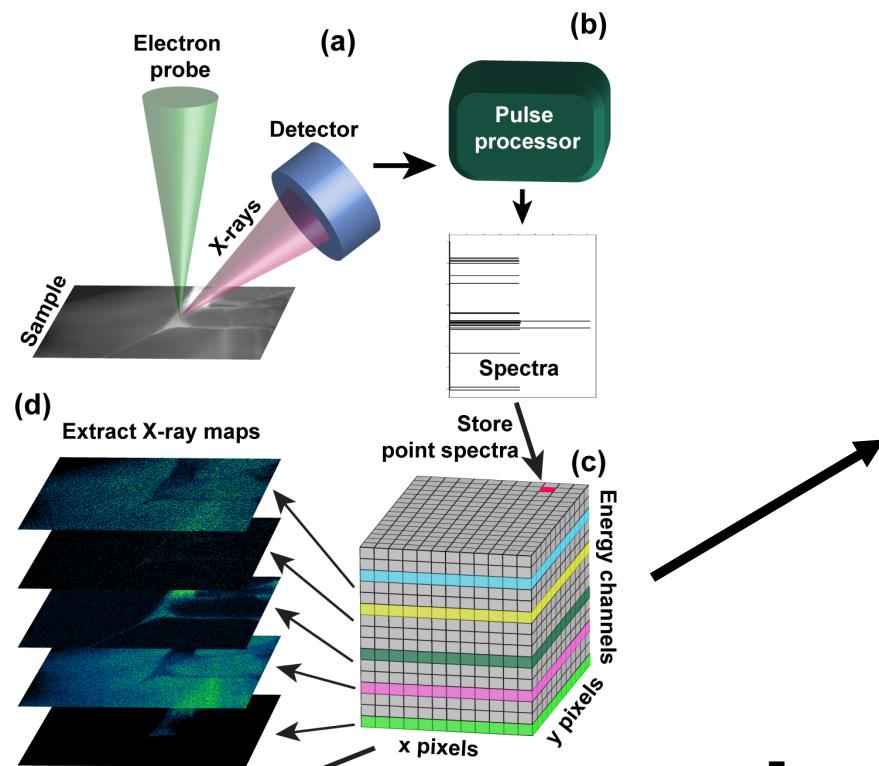
$$D \approx AS^T = TP^T$$

$$D \approx (TR)(R^{-1}P)^T = \tilde{T}\tilde{P}^T$$

# Turn raw data into a matrix

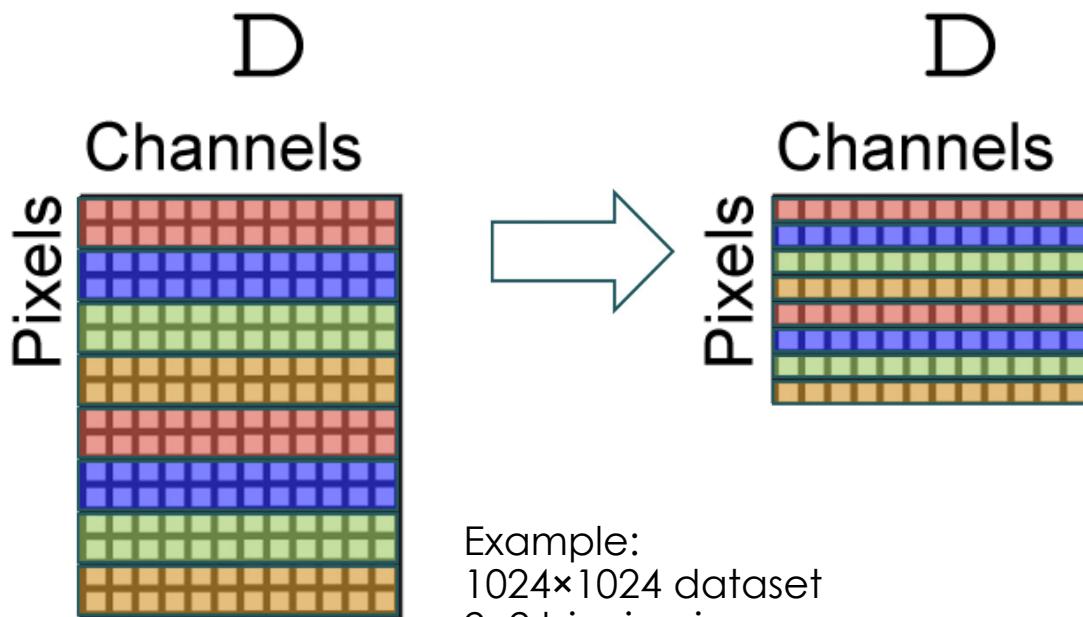
$$I(x, y|E) \rightarrow D_{n \times m}$$

Typically convert datatype to sparse here



Easy enough to reshape back to maps (x,y) later

## Optional: binning



Example:

1024×1024 dataset

2×2 binning in space

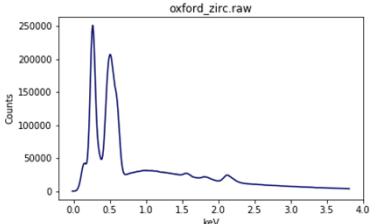
→ 512×512 data, 4× more counts/pixel, 2× degraded spatial resolution

Can also bin on energy (i.e., ×2) to increase counts as loss of energy resolution

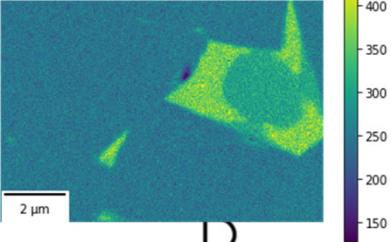
Easy sparse binning: Keenan et al., JVST A, V33(5) 2015, #05E123-1

# Need to account for Poisson noise

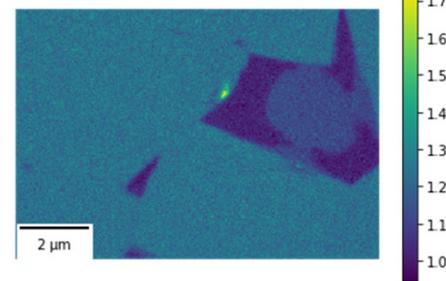
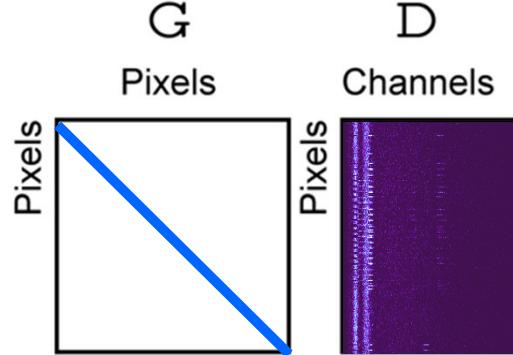
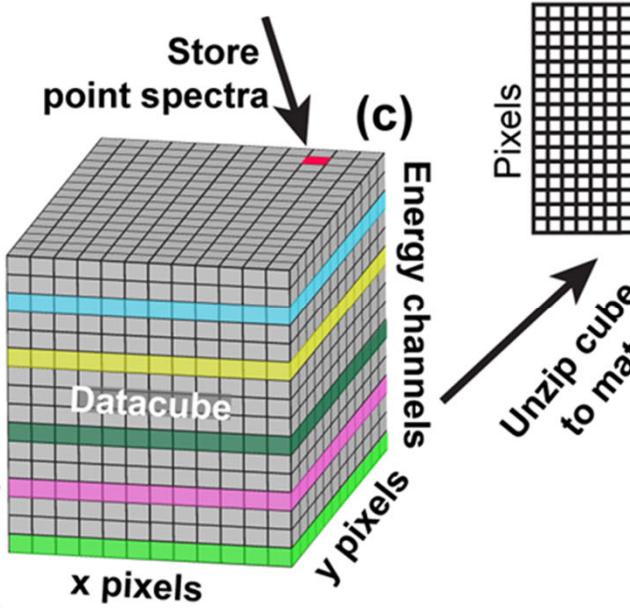
**Mean spectrum**



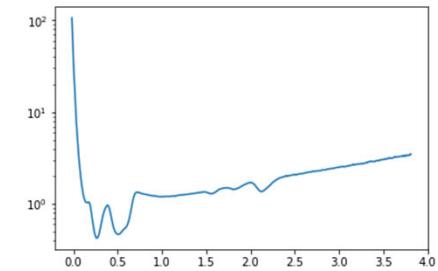
**Mean counts**



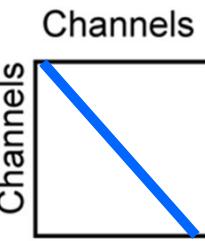
$$\hat{D} = GDH$$



$1 / \sqrt{(\text{mean spectrum})}$



H



Keenan, Tech and Apps.  
of Hyperspectral Imaging,  
Ch. 5, 2007

Keenan & Kotula, Surf.  
Int. Anal., V36(3) P. 203  
(2004)

Etc.

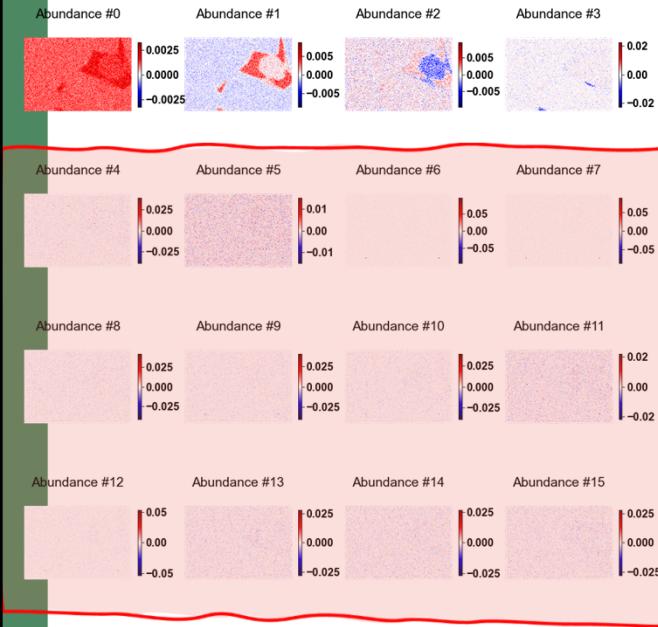
$1 / \sqrt{(\text{mean counts})}$

Recent insight:  
For very low count rates  
(i.e., nanoparticles in  
STEM),  $\mathbf{G}=1$  (spectral-  
only binning) may work  
better.

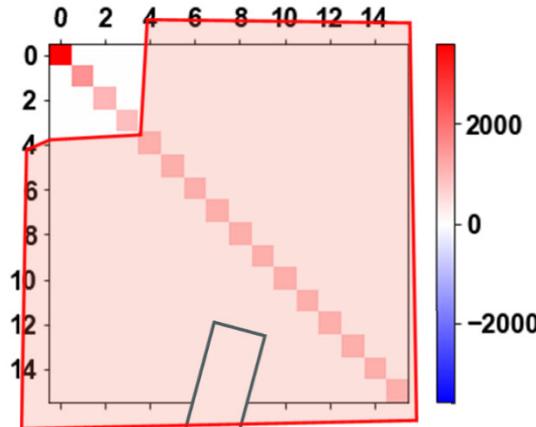
Kotula & Van Benthem  
(2015). M&M V21(S3):  
1423.

# Singular Value Decomposition

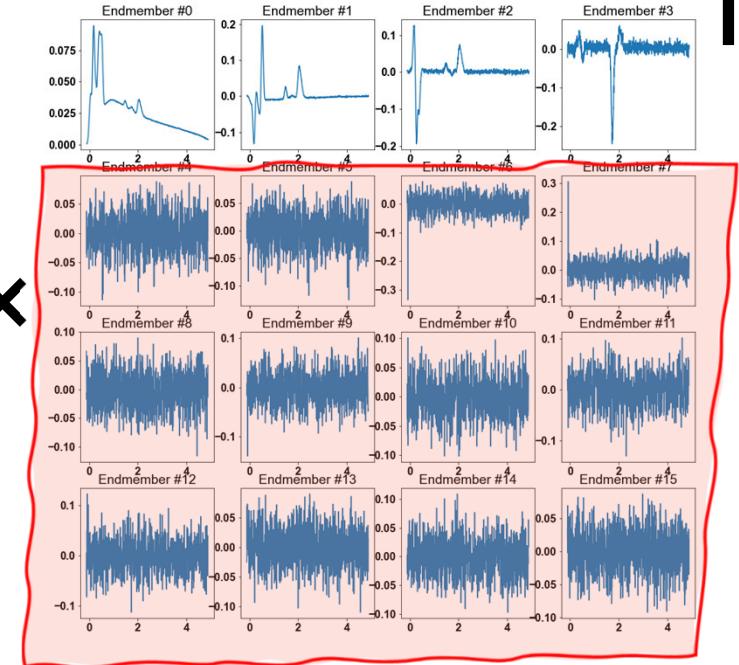
$$\hat{D} \approx U\Sigma V^T$$



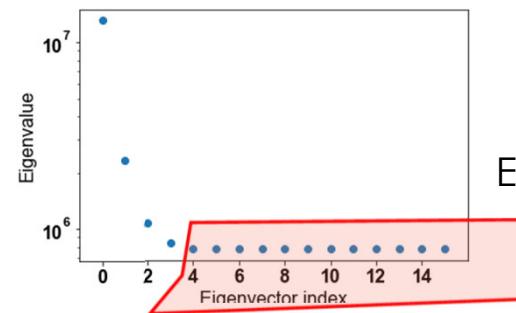
**X**



**X**



**X**



Eigenvalues = singular values<sup>2</sup>

Note: for this example,  
 $\mathbf{G}=1$ ,  
 $\mathbf{H}=1/\text{sqrt}(\text{mean spectrum})$   
i.e., spectral only

## Singular value decomposition $\Leftrightarrow$ principal component analysis

$$\hat{D} \approx U\Sigma V^T = \hat{A}\hat{S}^T$$

$$U\Sigma = \hat{A}, \quad V = \hat{S}$$

Or,

$$U = \hat{A}, \quad \Sigma V = \hat{S}$$

Then, return from Poisson-scaled space to “real” space:

$$S = H^{-1}\hat{S}$$

$$A = G^{-1}\hat{A}$$

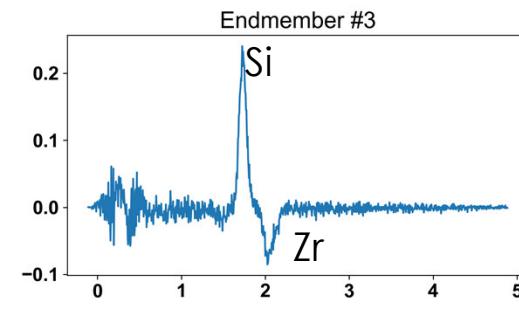
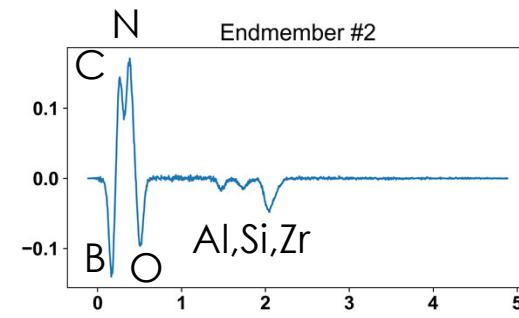
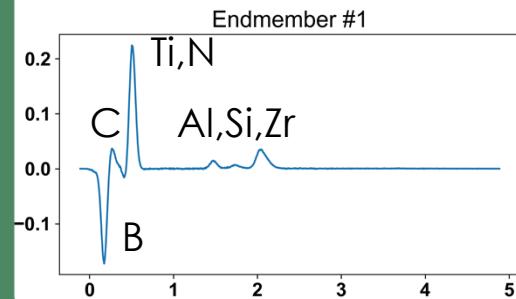
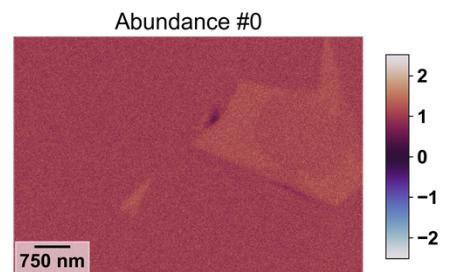
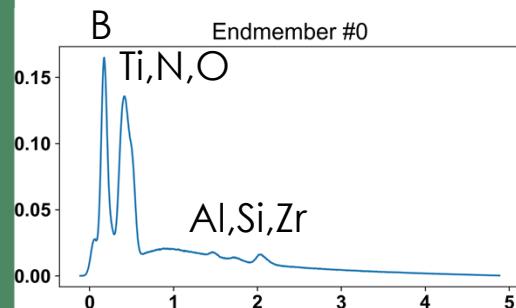
Important point: U, V are orthonormal, therefore A-hat, S-hat are orthonormal or orthogonal. But, the H<sup>-1</sup>, G<sup>-1</sup> steps don't preserve this orthogonality, so A, S are not orthogonal; this will be important later. Keenan (2007) shows a simple procedure “fPCA” to make the real-space, PCA-like model orthogonal: **Abundances + endmembers!!!!**

$$D \approx AS^T = TP^T$$

AS<sup>T</sup>: arbitrary factor model  
TP<sup>T</sup>: orthogonal factor model

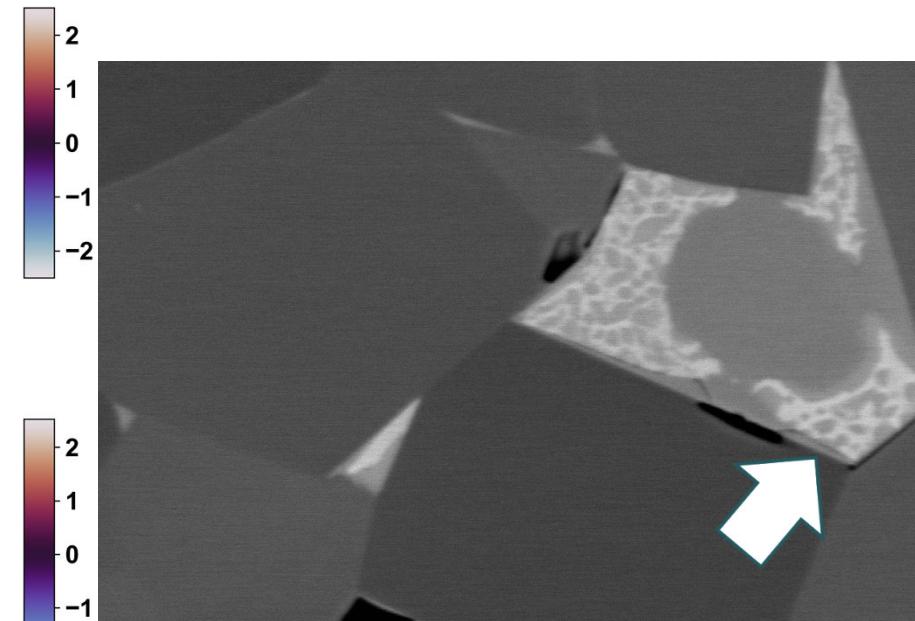
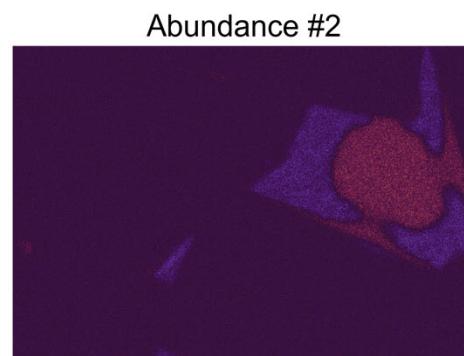
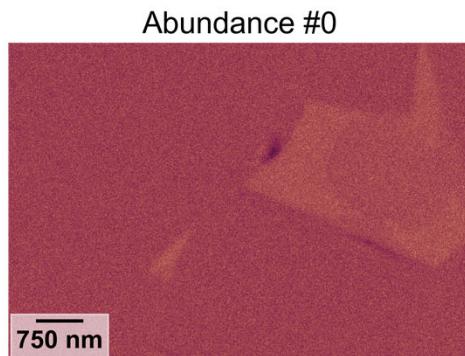
- Keenan, M. R. (2007). Techniques and Applications of Hyperspectral Image Analysis.  
H. F. Grahn and P. Geladi. Chichester, John Wiley & Sons: 89-126.  
Keenan, M. R. (2009). Surface and Interface Analysis 41: 79-87.  
Keenan, M. R. and P. G. Kotula (2004). Surface and Interface Analysis 36(3): 203-212.

# Abundances + endmembers

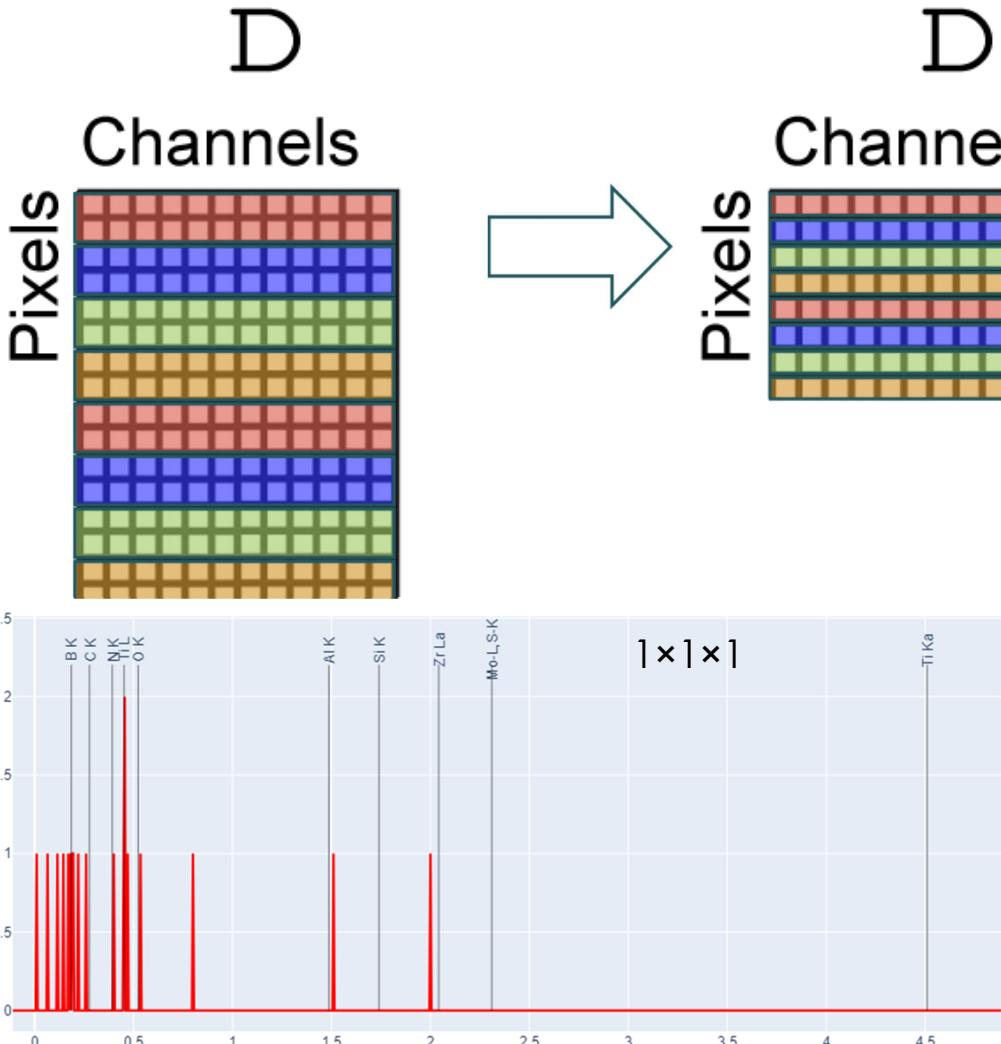


# Always ask yourself, “Does this make sense?”

- Machine learning methods can't help you if you don't bring your domain expertise to bear!



# Optional: binning

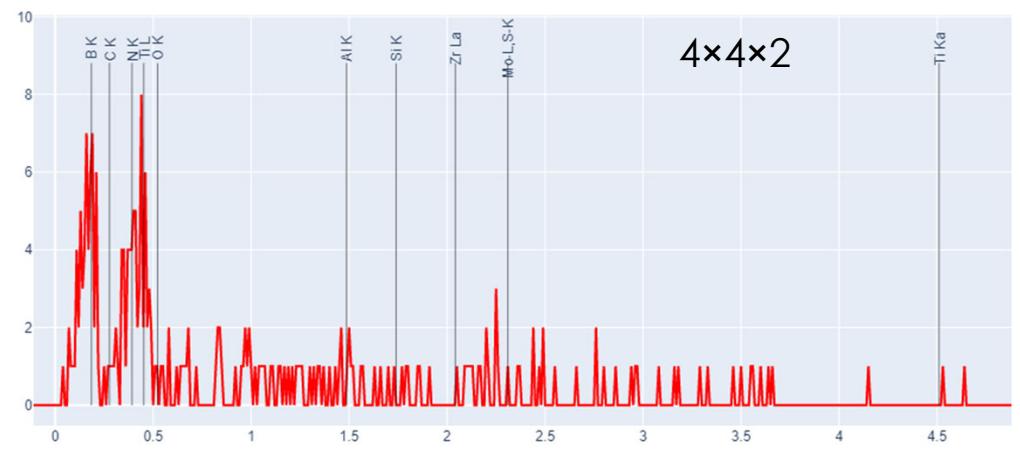


D

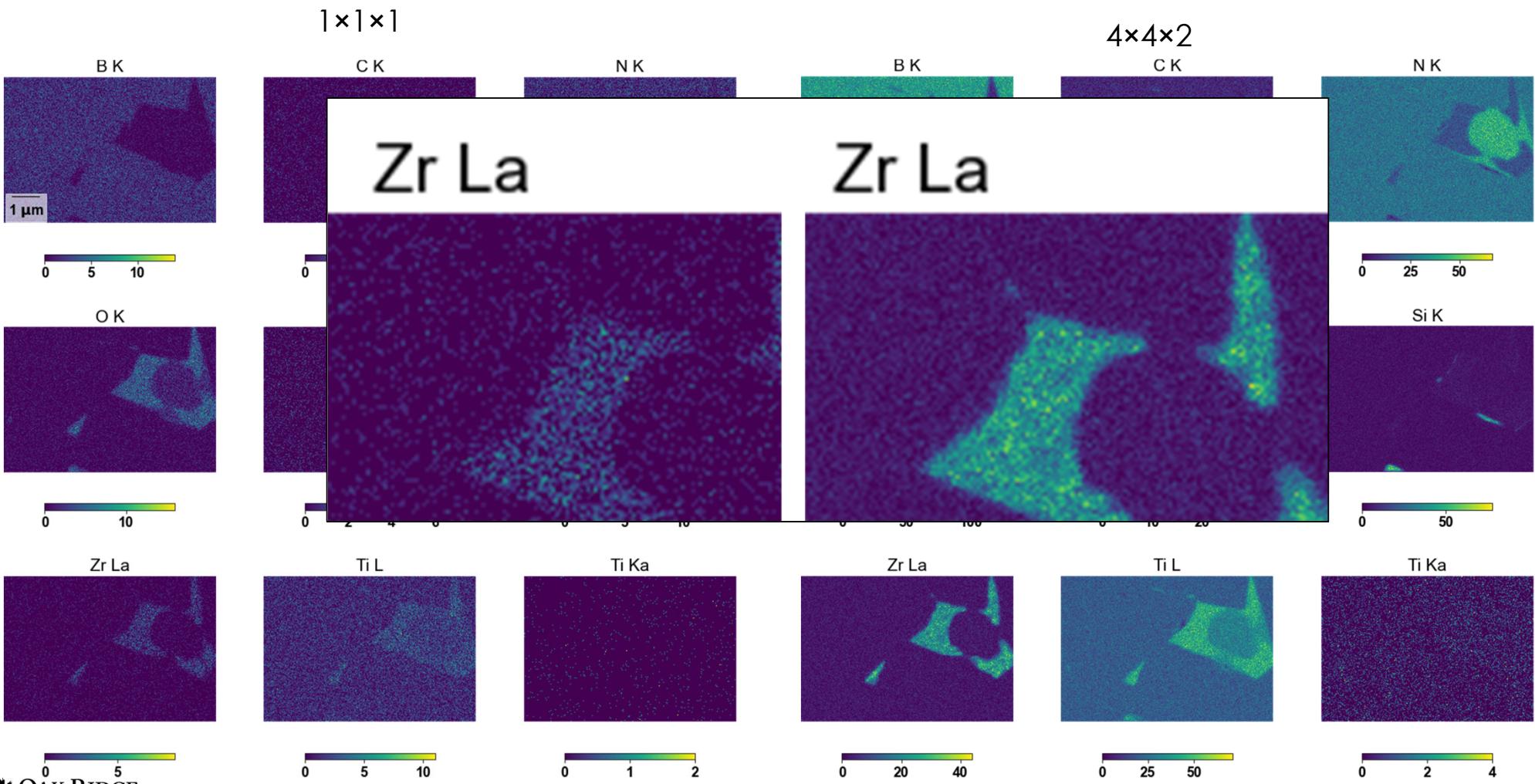
Channels

Pixels

Here: 4x4 space, ×2 spectral  
(32× aggregate binning)



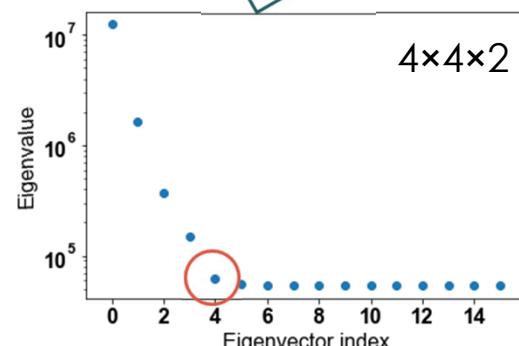
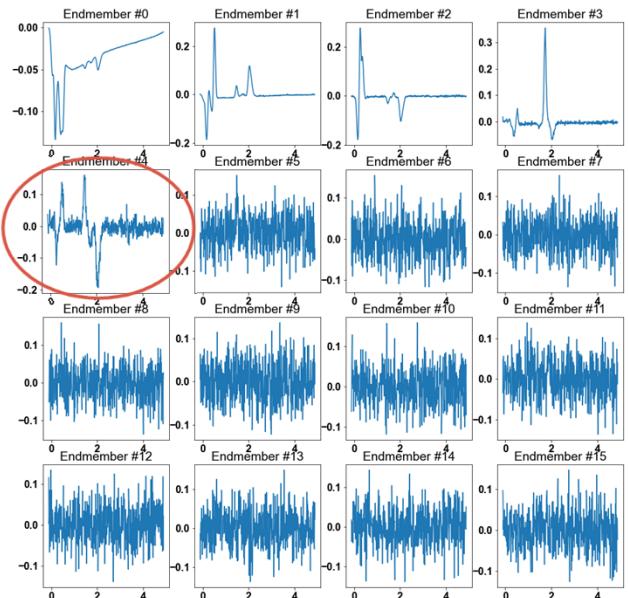
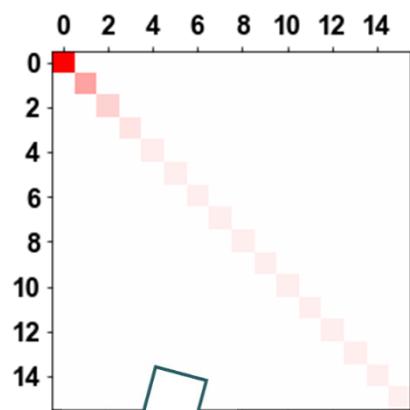
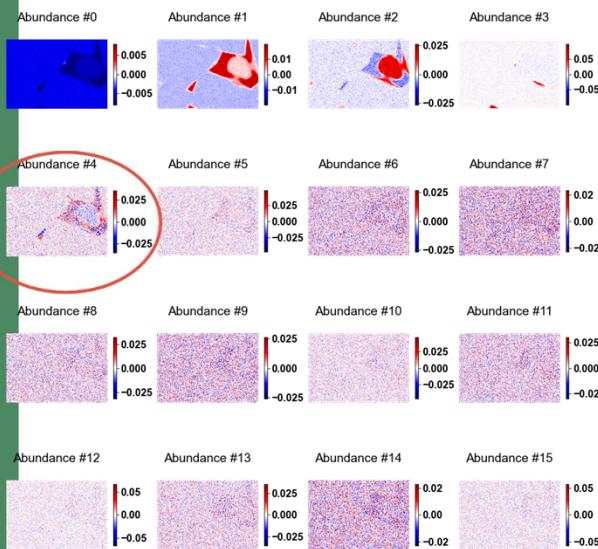
# Binning comparison



# Singular Value Decomposition: binned

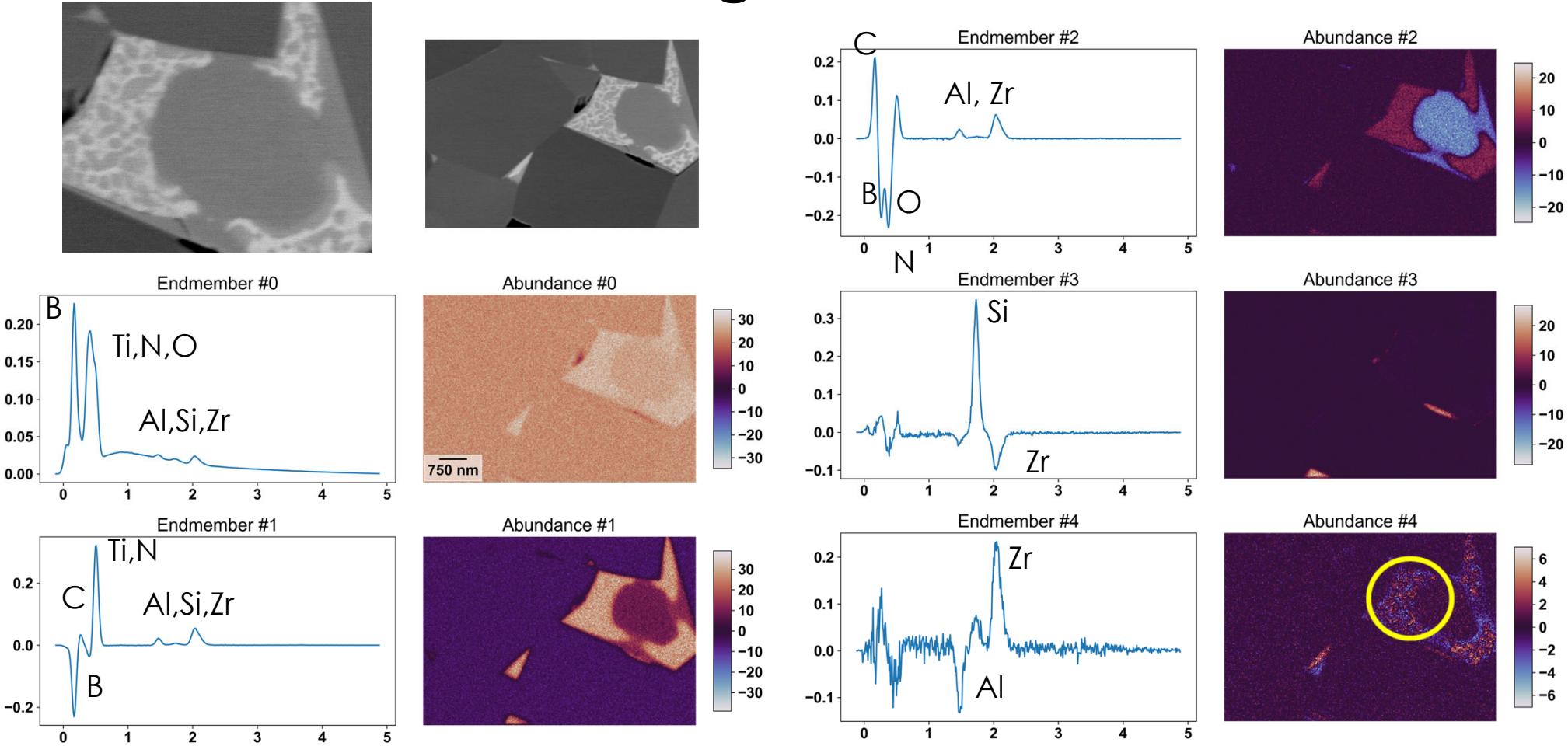
$$\hat{D} \approx U\Sigma V^T$$

T



Eigenvalues = singular values<sup>2</sup>

# Binned: now we're seeing the features!

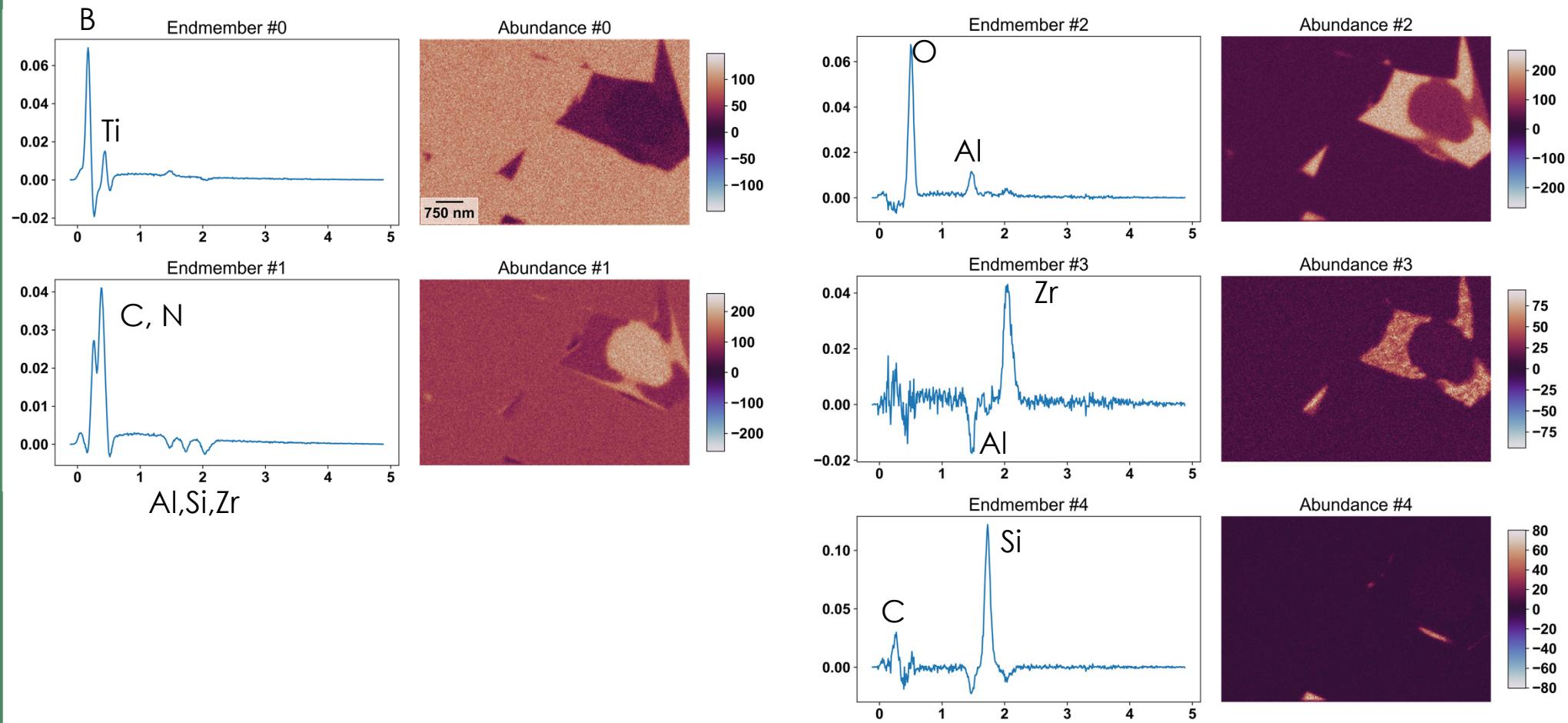


# Final step: rotations

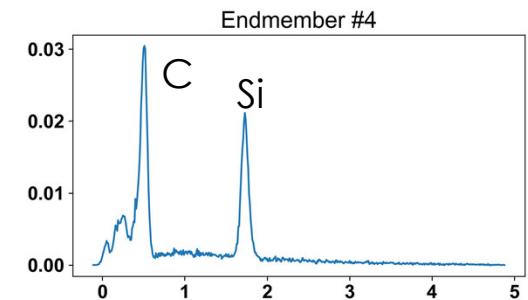
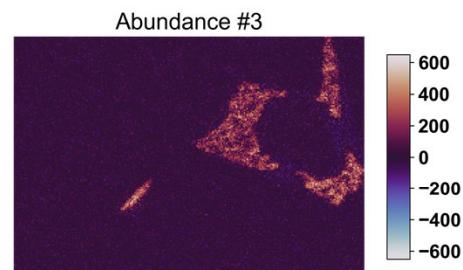
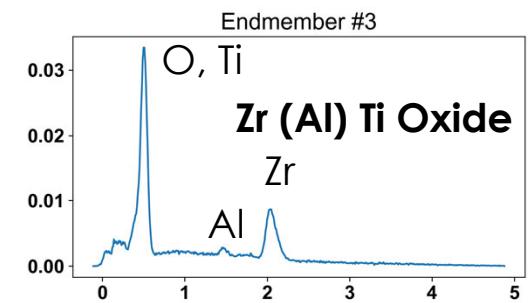
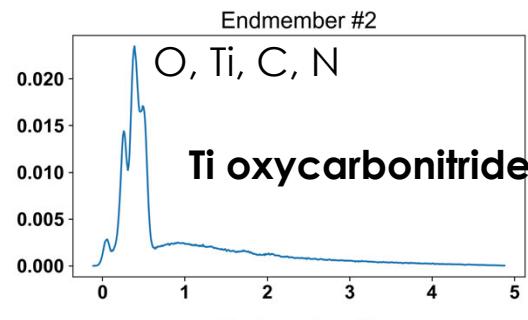
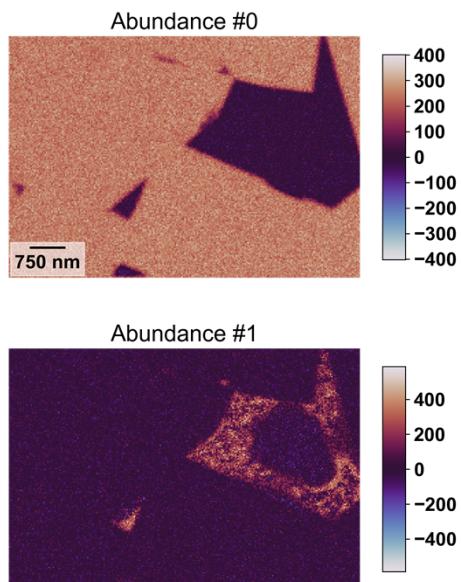
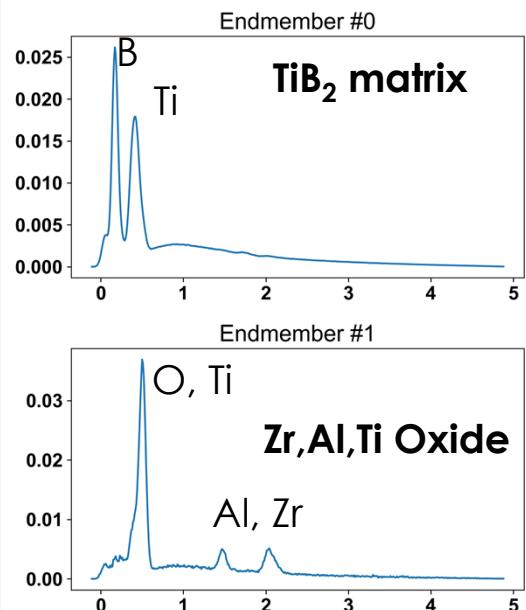
$$D \approx (TR)(R^{-1}P)^T = \tilde{T}\tilde{P}^T \quad (\text{Assuming } R^{-1} \text{ exists})$$

- **The goal here is to choose an “intelligent” R to make the abundances and endmembers easier to interpret**
- **Literally infinite R exist**
- **Generally choose *orthogonal* rotation,  $R^T=R^{-1}$**
- **Most common approach: “varimax criterion”**
- **Rotate to *simplicity*:**
  - “Simplicity” means, essentially, try to have only one strong value at each pixels or channel
  - Make “P” simple → “spectral simplicity”
  - Make “T” simple → “spatial simplicity”
- **Simple concept, but very pedantic in detail:**
  - Smentkowski, V. S., S. G. Ostrowski and M. R. Keenan (2009). Surface and Interface Analysis 41: 88-96.
  - Keenan, M. R. (2007). Techniques and Applications of Hyperspectral Image Analysis. H. F. Grahn and P. Geladi. Chichester, John Wiley & Sons: 89-126.
  - Keenan, M. R. (2009). Surface and Interface Analysis 41: 79-87. 

# Spectral simplicity: find elements



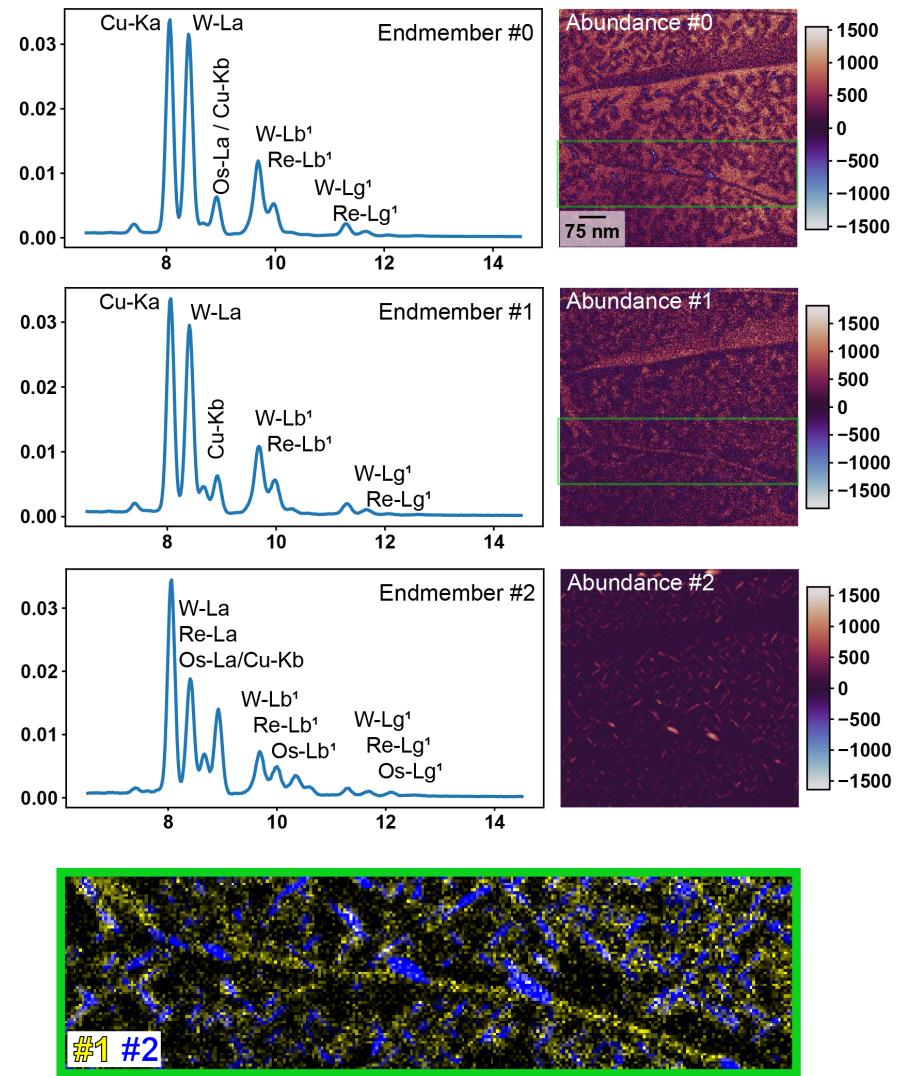
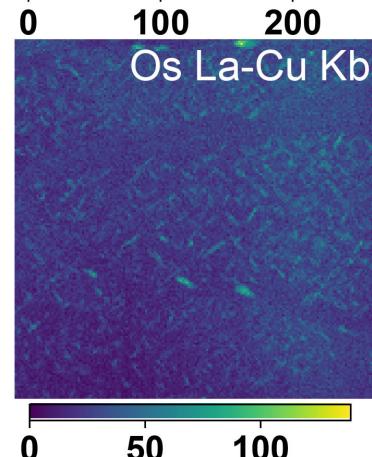
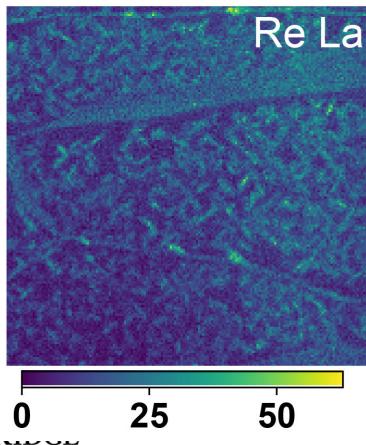
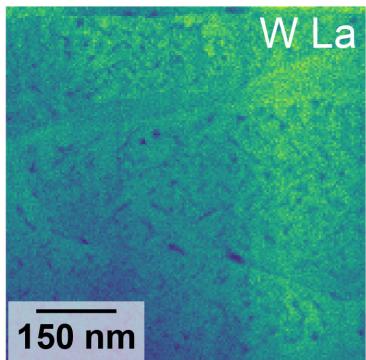
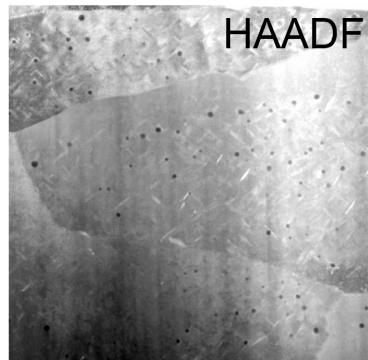
# Spectral simplicity: find “phases”



**Polishing media (colloidal silica)**

# Another example: STEM EDS XSI

Neutron-irradiated tungsten,  
W-Re-Os alloy (Cu TEM holder)



## **Topics not covered today:**

- **Cluster analysis (K-means, etc.)**
- **Choosing analysis energy ranges**
- **Pros and cons of binning**
- **Blind source separation / independent component analysis**
- **Combining with other signals (i.e., EBSD)**
- **Non-linear effects: absorption, fluorescence**

**Feel free to contact me offline: [parishcm@ornl.gov](mailto:parishcm@ornl.gov)**

## **Machine Learning of X-ray spectrum images**

- **S(T)EM-EDS XSIs are large, noisy, and sparse, and therefore ideal for computational attack**
- **Unsupervised approaches seem most suitable**
- **The physics are close to linear, so bilinear factor analysis (i.e., PCA/SVD) is supremely suited**
- **Interpretation of the results requires domain expertise**