

# Photocatalytic Conversion of CO<sub>2</sub> into Renewable Hydrocarbon Fuels: State-of-the-Art Accomplishment, Challenges, and Prospects

Wenguang Tu, Yong Zhou,\* and Zhigang Zou\*

Photocatalytic reduction of CO<sub>2</sub> into hydrocarbon fuels, an artificial photosynthesis, is based on the simulation of natural photosynthesis in green plants, whereby O<sub>2</sub> and carbohydrates are produced from H<sub>2</sub>O and CO<sub>2</sub> using sunlight as an energy source. It couples the reductive half-reaction of CO<sub>2</sub> fixation with a matched oxidative half-reaction such as water oxidation, to achieve a carbon neutral cycle, which is like killing two birds with one stone in terms of saving the environment and supplying future energy. The present review provides an overview and highlights recent state-of-the-art accomplishments of overcoming the drawback of low photoconversion efficiency and selectivity through the design of highly active photocatalysts from the point of adsorption of reactants, charge separation and transport, light harvesting, and CO<sub>2</sub> activation. It specifically includes: i) band-structure engineering, ii) nano-structuralization, iii) surface oxygen vacancy engineering, iv) macro-/meso-/microporous structuralization, v) exposed facet engineering, vi) co-catalysts, vii) the development of a Z-scheme system. The challenges and prospects for future development of this field are also present.

photosystem I (PSI).<sup>[1]</sup> The idea of mimicking the overall natural photosynthetic cycle of chemical conversion of CO<sub>2</sub> into hydrocarbon fuels has been consistently drawing attention for more than 30 years, including biological conversion,<sup>[1]</sup> thermochemical conversion,<sup>[2]</sup> electrochemical conversion,<sup>[3]</sup> and photocatalytic reduction of CO<sub>2</sub>.<sup>[4]</sup> During the conversion process, a high input of energy is needed for breaking the C=O bond due to the thermodynamic stability of CO<sub>2</sub>. Compared with other methods, photocatalytic reduction of CO<sub>2</sub> into hydrocarbon fuels is a challenging, yet promising avenue for achieving a sustainable alternative to conventional fossil fuels with advantages as follows: i) it can be carried out in relatively mild conditions – room temperature and pressure; ii) this process uses a mass of abandoned CO<sub>2</sub> as the starting carbon source driven by inexhaustible and clean solar energy; iii) the

## 1. Introduction

Photosynthesis uses solar light energy to couple the formation of O<sub>2</sub> and carbohydrates with the fixation of CO<sub>2</sub>. This process produces a readily usable carbon source and an aerobic atmosphere to sustain almost all life on this planet. The center of this process is photosystem II (PSII), which photoinduces the water oxidation that provides a primary source of reducing equivalents (water-derived electrons and protons) to convert CO<sub>2</sub> into biomass, food, and fuel with an additional input of energy from

photoreduction of CO<sub>2</sub> can directly generate short-chain hydrocarbon fuels such as CH<sub>4</sub>, CH<sub>3</sub>OH, C<sub>2</sub>H<sub>6</sub> and so on, which can alleviate increasingly tense energy crisis; iv) the realization of this technology will make it possible to replace fossil fuels by CO<sub>2</sub> as C source in chemical industry. Therefore, photocatalytic reduction of CO<sub>2</sub> with H<sub>2</sub>O into hydrocarbon fuels would be like killing two birds with one stone in terms of saving supplying energy and our environment.<sup>[5]</sup>

In spite of earlier reviews generally introducing the concept of artificial photosynthesis,<sup>[6–20]</sup> the present review beginning with a brief description of the basic principles of photocatalytic reduction of CO<sub>2</sub> provides an overview and highlights the state-of-the-art accomplishments of overcoming the drawback of low photoconversion efficiency and selectivity through the design of highly active photocatalysts from the point of view of adsorption of reactants, charge separation and transport, light harvesting, and CO<sub>2</sub> activation. It includes: i) band-structure engineering, an effective approach to adjust bandgap and band position for utilizing solar energy effectively; ii) nanostructuralization, providing a direct pathway for electron transport, which is favorable for improving separation of photogenerated electron–hole pairs; iii) surface oxygen vacancy engineering, which is favorable for CO<sub>2</sub> adsorption and activation and changes the electronic and chemical properties of the semiconductor surfaces; iv) macro/meso/microporous structuralization, which results in readily access channel, and the large surface area that

Dr. W. Tu, Prof. Y. Zhou  
Key Laboratory of Modern Acoustics  
MOE, Institute of Acoustics  
School of Physics  
Nanjing University  
22 Hankou Road, Nanjing, Jiangsu, 210093, P. R. China  
E-mail: zhouyong1999@nju.edu.cn  
Dr. W. Tu, Prof. Y. Zhou, Prof. Z. Zou  
National Laboratory of Solid State Microstructures  
School of Physics  
Ecomaterials and Renewable Energy Research Center (ERERC)  
Jiangsu Key Laboratory for Nano Technology  
Nanjing University  
22 Hankou Road, Nanjing, Jiangsu 210093, P. R. China  
E-mail: zgrou@nju.edu.cn

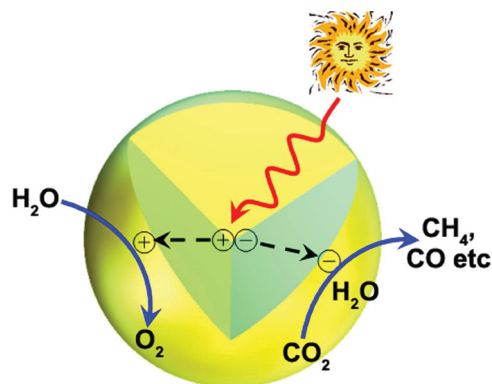


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increases the adsorption of reactants and supplies more surface active sites; v) exposed facet engineering, which could adjust surface energy, surface active sites, and even electronic band structure; vi) co-catalysts, which can promote the separation and migration of photogenerated charge carriers, effectively lower the barrier for  $\text{CO}_2$  activation, affect the selectivity of products, and inhibit the back-recombination reaction; vii) the development of a Z-scheme system that relies on both exploration of an efficient semiconductor photocatalyst and creation of active sites to promote surface chemical reactions in the presence of a redox mediator, and simultaneously suppress backward reactions involving redox mediator. The challenges and prospects for future development of this field are also present.

## 2. Basic Principles of Photocatalytic Reduction of $\text{CO}_2$

$\text{CO}_2$ , a linear molecule, is one of most thermodynamically stable compounds of carbon. Photocatalytic reduction of  $\text{CO}_2$  into hydrocarbon fuels demands input energy to break  $\text{C}=\text{O}$  bond and form  $\text{C}-\text{H}$  bond, involving the participation of multiple electrons and a corresponding number of protons.<sup>[21]</sup> Due to the highest chemical state ( $\text{C}^{4+}$ ) of C atoms in  $\text{CO}_2$ ,  $\text{CO}_2$  can only be reduced with the support of reducing agents.  $\text{H}_2\text{O}$  is a preferred candidate of reducing agents compared with other ones such as  $\text{H}_2$ ,  $\text{S}^{2-}$ ,  $\text{SO}_3^{2-}$ , and amines, owing to its richness, non-toxicity, and effectiveness. Photocatalytic reduction of  $\text{CO}_2$  with  $\text{H}_2\text{O}$  into hydrocarbon fuels such as  $\text{CH}_4$  and  $\text{CH}_3\text{OH}$  is an uphill reaction with a highly positive change in Gibbs free energy:  $\text{CO}_2 + 2\text{H}_2\text{O} \rightarrow \text{CH}_3\text{OH} + 3/2\text{O}_2$  ( $\Delta G^\circ = 702.2 \text{ kJ mol}^{-1}$ ) and  $\text{CO}_2 + 2\text{H}_2\text{O} \rightarrow \text{CH}_4 + 2\text{O}_2$  ( $\Delta G^\circ = 818.3 \text{ kJ mol}^{-1}$ ). Thus, the input energy is used to overcome these reaction barriers, which is provided by incident light. The schematic illustration of basic mechanism of photocatalytic reduction of  $\text{CO}_2$  is shown in Figure 1. In the photocatalytic processes, electrons are excited from the valence band (VB) to conduction band (CB) and an equal numbers of holes are generated in the CB simultaneously when semiconductor absorbs a flux of photons. The photogenerated electron-hole pairs separate from each other and vigorously move to catalytically active sites at semiconductor surface. At the surface of semiconductors, electrons



**Figure 1.** Schematic illustration of photoinduced generation of an electron-hole pair in semiconductor that transfers to the surface for  $\text{CO}_2$  photoredox.



nanomaterials for photocatalysis application.

**Wenguang Tu** obtained his BS degree in 2010 from Southwest University, China. He is pursuing his Ph.D. degree under the guidance of Prof. Yong Zhou in ERERC, School of Physics, National Laboratory of Solid State Microstructures, Nanjing University, China. His current research involves the synthesis of graphene-based



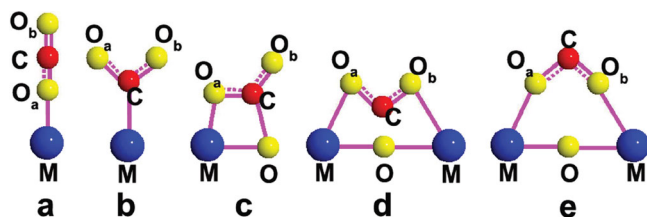
2003–2004, the National Institute of Advanced Industrial Science and Technology (AIST) (Japan) in 2004–2008, and the National University of Singapore (NUS) in 2008–2009, he joined as a full professor in the Eco-materials and Renewable Energy Research Center (ERERC), School of Physics, National Laboratory of Solid State Microstructures, Nanjing University, China. His research now focuses on the design and fabrication of solar-light-driven clean energy materials for photocatalysis and flexible solar cells.

**Yong Zhou** studied chemistry and physics at the University of Science and Technology of China (USTC), and received his Ph.D. degree there in 2000. After working in Kyoto University in 2000–2001, the Max Planck Institute of Colloids and Interfaces in 2002–2003, the National Institute of Materials Science (NIMS) (Japan) in



Institute of Technology (Japan) as a visiting researcher. After he had received his Ph.D. degree from the University of Tokyo (Japan) in 1996, he was a researcher at the Photoreaction Control Research Center, National Institute of Advanced Industrial Science and Technology (AIST), Japan. He has been with the Department of Physics at Nanjing University (China) as a distinguished professor of the Chang Jiang Scholars Program since 2003. He is also a Director of the Ecomaterials and Renewable Energy Research Center, Nanjing University. His current research interests are photocatalysis, solar cells, and fuel cells.

**Zhigang Zou** received his bachelors degree and masters degree from Tianjing University (China) in 1982 and 1986, respectively. He was a lecturer of Tianjing University from September 1986 to March 1991. Then, he moved to the Research Laboratory of Engineering Materials at the Tokyo

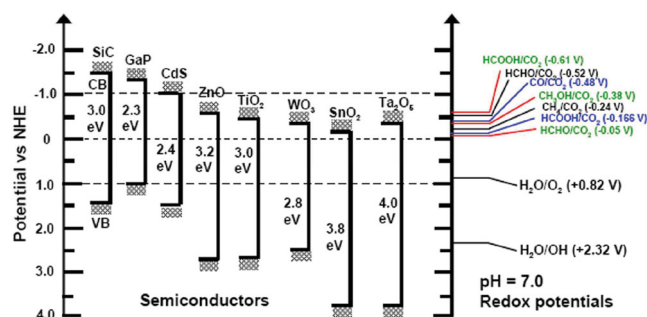


**Figure 2.** Possible configurations of adsorbed CO<sub>2</sub> on the photocatalyst surface.

with reducing power reduce CO<sub>2</sub> into hydrocarbon fuels such as CO, CH<sub>3</sub>OH, and CH<sub>4</sub> in the presence of H<sub>2</sub>O, while the oxidation of H<sub>2</sub>O occurs, owing to the oxidation ability of holes.<sup>[22]</sup> Through the chemical transformations, the abundant solar energy in the world can thus be effectively converted into chemical energy.

It was very essential to first understand the initial CO<sub>2</sub> absorption and activation mechanism so that effective photocatalysts could be devised to promote CO<sub>2</sub> reduction efficiency.<sup>[23,24]</sup> Five models are constructed for CO<sub>2</sub> absorption, and different absorption model determines the different adsorption energy of the system (Figure 2).<sup>[25]</sup> The first one is that the CO<sub>2</sub> molecule is linearly adsorbed on the surface via the O<sub>a</sub> atom (Figure 2a). The second is that CO<sub>2</sub> molecule is absorbed via the C atom to generate a monodentate carbonate species (Figure 2b). In the third, a bidentate carbonate species is generated through interaction of a CO<sub>2</sub> molecule with the surface via both the O<sub>a</sub> and C atom (Figure 2c). The fourth is the generation of a bridged carbonate geometry with the C atom of CO<sub>2</sub> pointing downward and two O atoms of CO<sub>2</sub> binding with two metal atoms to form a C···O bond with the O atom on the surface (Figure 2d). In the fifth, a bridging configuration with the C atom of CO<sub>2</sub> pointing upward and two O atoms of CO<sub>2</sub> binding with two metal atoms is formed (Figure 2e). The presence of an M–O–M bond on the surface is contributed to the formation of the fourth or the fifth model.

The activation of chemically stable CO<sub>2</sub> initiates multistep reactions for CO<sub>2</sub> reduction, which likely involves one electron transfer to CO<sub>2</sub> to generate CO<sub>2</sub><sup>•−</sup> species by transferring an electron from excited photocatalysts to the lowest unoccupied molecular orbital (LUMO) of CO<sub>2</sub>.<sup>[10,26–29]</sup> Accompanied with the acceptance of one electron, the transformation of CO<sub>2</sub> geometry from linear structure to a bent form also occurs significantly, owing to the repulsion among the two lone electron pairs on the oxygen atoms and the unpaired electron on the carbon atom.<sup>[30]</sup> The LUMO of CO<sub>2</sub> is lowered in energy upon decreasing the O–C–O bond angle from 180°. The non-linear CO<sub>2</sub> molecule generated on the surface of solid bases is more destabilized than the linear CO<sub>2</sub> molecule, showing the high reactivity for CO<sub>2</sub> photoreduction.<sup>[31,32]</sup> Activated CO<sub>2</sub> is probably an important species in the photochemical reactions of CO<sub>2</sub>. The transfer of one electron to free CO<sub>2</sub> is unfavorable thermodynamically, because the high LUMO level of CO<sub>2</sub> leads to a very negative redox potential for this process of CO<sub>2</sub> + e<sup>−</sup> → CO<sub>2</sub><sup>•−</sup> (E<sub>redox</sub><sup>0</sup> = −1.90 V, vs NHE), which has been proved by scanning tunneling microscopy experiments.<sup>[28,29]</sup> Thus, multiple electrons and a corresponding number of protons must involve the chemical reactions, reflecting the multicomponent



**Figure 3.** Conduction band, valence band potentials, and band gap energies of various semiconductor photocatalysts relative to the redox potentials at pH 7 of compounds involved in CO<sub>2</sub> reduction.

steps observed in natural photosynthetic systems.<sup>[33]</sup> The formation of products can be different, decided by the number of electrons and protons taking part in the chemical reaction. CO is formed by reacting with two protons and two electrons, while CH<sub>4</sub> formation needs eight electrons and eight protons. The selectivity of products is one of considerable problems in the photocatalytic reduction of CO<sub>2</sub> process, which may be influenced by photocatalysts, reaction condition, and thermodynamic reduction potentials. Figure 3 shows a schematic of different semiconductor photocatalysts relative to the redox potentials of different species. Two plausible pathways for CH<sub>4</sub> formation are proposed: i) CO<sub>2</sub> → HCOOH → HCHO → CH<sub>3</sub>OH → CH<sub>4</sub>; and ii) CO<sub>2</sub> → CO → C<sup>•</sup> → CH<sub>2</sub> → CH<sub>4</sub>.<sup>[13,14]</sup> However, the understanding of photocatalytic mechanism and selectivity is still in infant due to complex multielectron-transfer process.<sup>[8,10,12]</sup>

### 3. State-of-the-Art Accomplishments of Design and Engineering of Highly Active Photocatalysts

Reducing CO<sub>2</sub> with photon energy requires materials that efficiently absorb light to generate electron–hole pairs with the eventual target of achieving a carbon-neutral artificial photosynthesis cycle by coupling the reductive half-reaction of CO<sub>2</sub> fixation with a matched oxidative half-reaction such as water oxidation. To achieve the photocatalytic reduction of CO<sub>2</sub> thermodynamically, the bottom of CB of photocatalysts should have a more negative potential than CO<sub>2</sub> reduction potentials, and the top of VB should be located at a more positive potential than the H<sub>2</sub>O oxidation potential. The efficient performance of semiconductor photocatalysts should benefit from: i) the bandgap of semiconductors causes absorption of incident light to generate the electron–hole pairs across the region of solar spectrum; ii) electron–hole pairs separate effectively from each other and migrate to the surface; iii) photocatalysts provides enough quality and quantity of active sites such as adsorption sites and reaction centres, contributing to photocatalytic reaction. The overall efficiency of photocatalytic reduction of CO<sub>2</sub> is determined by the balance of thermodynamics and kinetics of the processes above.

In 1979, Inoue et al. initially reported photocatalytic reduction of CO<sub>2</sub> to organic compounds on various semiconductor photocatalysts suspended in CO<sub>2</sub> saturated water illuminated by



a Xe lamp, and different products have been obtained, such as CO, CH<sub>4</sub>, CH<sub>3</sub>OH, HCOOH or others.<sup>[4]</sup> Since photocatalysis is a surface/interface-reaction process, catalysts play an important role in development of useful transformations utilizing this attractive and environmentally friendly raw material. A series of characteristic properties of photocatalysts have a direct impact on the CO<sub>2</sub> conversion efficiency. So far, explored semiconductor materials for photocatalytic reduction of CO<sub>2</sub> include, but not limited to, metal oxides, metal sulfides, nitrides, and phosphides, as summarized in **Table 1**. Most metal oxides consist of metal cations with d<sup>0</sup> (Ti<sup>4+</sup>, Zr<sup>4+</sup>, Nb<sup>5+</sup>, Ta<sup>5+</sup>, V<sup>5+</sup>, Mo<sup>6+</sup>, W<sup>6+</sup>) and d<sup>10</sup> (In<sup>3+</sup>, Ga<sup>3+</sup>, Ge<sup>4+</sup>, Sn<sup>4+</sup>, Sb<sup>5+</sup>) configurations. The d<sup>0</sup> metal oxides contain titanates [TiO<sub>2</sub>,<sup>[4,10,26,34–149]</sup> ATiO<sub>3</sub> (A = Sr, Ca, Ba, Pb),<sup>[150–154]</sup> K<sub>2</sub>Ti<sub>6</sub>O<sub>13</sub>,<sup>[155,156]</sup> Al<sub>2</sub>Ti<sub>4</sub>O<sub>15</sub> (A = Ca, Sr, Ba),<sup>[157]</sup> ZrO<sub>2</sub>,<sup>[158–161]</sup> niobates [HfNb<sub>3</sub>O<sub>8</sub>,<sup>[162,163]</sup> InNbO<sub>4</sub>,<sup>[164]</sup> ANbO<sub>3</sub> (A = Li, Na, K),<sup>[165–169]</sup> tantalates [Ta<sub>2</sub>O<sub>5</sub>,<sup>[170,171]</sup> InTaO<sub>4</sub>,<sup>[172–176]</sup> ATaO<sub>3</sub> (A = Li, Na, K),<sup>[177]</sup> vanadates (BiVO<sub>4</sub>,<sup>[178]</sup> Fe<sub>2</sub>V<sub>4</sub>O<sub>13</sub>,<sup>[179]</sup> Na<sub>2</sub>V<sub>6</sub>O<sub>16</sub>,<sup>[180]</sup> Bi<sub>6</sub>Mo<sub>2</sub>O<sub>15</sub>,<sup>[181]</sup> and tungstates (WO<sub>3</sub>,<sup>[182,183]</sup> W<sub>18</sub>O<sub>49</sub>,<sup>[184]</sup> Bi<sub>2</sub>WO<sub>6</sub>,<sup>[185,186]</sup> d<sup>10</sup> metal oxides includes In(OH)<sub>3</sub>,<sup>[187]</sup> gallates (Ga<sub>2</sub>O<sub>3</sub>,<sup>[188–190]</sup> ZnGa<sub>2</sub>O<sub>4</sub>,<sup>[191,192]</sup> CuGaO<sub>2</sub>,<sup>[193]</sup> and germinates (Zn<sub>2</sub>GeO<sub>4</sub>,<sup>[194–197]</sup> In<sub>2</sub>Ge<sub>2</sub>O<sub>7</sub>,<sup>[198]</sup> and stannates (Zn<sub>2</sub>SnO<sub>4</sub>,<sup>[199]</sup> Furthermore, d<sup>10</sup> metal nitride solid solutions such as (Zn<sub>1+x</sub>Ge)(N<sub>2</sub>O<sub>x</sub>)<sup>[200,201]</sup> and (Ga<sub>1-x</sub>Zn<sub>x</sub>)(N<sub>1-x</sub>O<sub>x</sub>)<sup>[202]</sup> were also found to be active. Metal sulfides such as CdS and ZnS,<sup>[4,203–214]</sup> MnS,<sup>[215]</sup> Cu<sub>x</sub>Ag<sub>y</sub>In<sub>z</sub>Zn<sub>k</sub>S solid solutions,<sup>[216]</sup> and Cu<sub>2</sub>ZnSnS<sub>4</sub><sup>[217]</sup> have been applied for photocatalytic reduction of CO<sub>2</sub> in various solvents. The p-type silicon,<sup>[218–220]</sup> p-type InP<sup>[221–223]</sup> and p-GaP<sup>[224]</sup> as photocathodes have successfully achieved photo-electrochemical reduction of CO<sub>2</sub>.

Although the study of the photocatalytic reduction of CO<sub>2</sub> started earlier, however, the CO<sub>2</sub> conversion efficiency still suffers from very low efficiencies. Fortunately, the design and engineering of photocatalysts with different architectures and surface properties have recently accelerated the development of novel and efficient photocatalyst, which involve bandgap engineering, nanostructuralization, surface oxygen vacancy engineering, macro/meso/microporous structuralization, exposure of highly active crystal facets, utilization of co-catalysts, and Z-scheme construction.

### 3.1. Band-Structure Engineering

Band-structure engineering is an effective approach to adjust bandgap and band position for utilizing solar energy effectively, and subsequently improve photoconversion efficiency of CO<sub>2</sub> reduction, involving quantum confinement effect, cation or anion doping, solid solution formation, hetero-structuralization and sensitization.

#### 3.1.1. Quantum Confinement Effect

Quantum-confined semiconductor nanocrystals (NCs) have many ideal characteristics for light-harvesting and charge-separation applications.<sup>[225–227]</sup> Compared with molecular chromophores,<sup>[228,229]</sup> NCs offer unique size-dependent absorption properties, large extinction coefficients over a broad spec-

tral range, the possibility of generating multiple excitons by single photons, long exaction lifetimes, and enhanced photostability.<sup>[230]</sup> The surface of NCs can be readily modified for specific functional targets and/or reaction environments.<sup>[231]</sup> Higher yields of CH<sub>3</sub>OH and CH<sub>4</sub> over the TiO<sub>2</sub> nanoparticles as the particle size decreased were detected under the illumination of light. The optimum particle size was a result of competing effects of specific surface area, charge-carrier dynamics, and light absorption efficiency.<sup>[82]</sup> Distinct from the bulk WO<sub>3</sub> for thermodynamically inactive CO<sub>2</sub> photoreduction, 4–5 nm thick WO<sub>3</sub> ultrathin nanosheets with dominant exposed {001} facets exhibited obvious performance for photocatalytic reduction of CO<sub>2</sub> into CH<sub>4</sub> under visible light irradiation, due to that size-quantization effects in vertical direction enlarges the WO<sub>3</sub> band gap.<sup>[182]</sup> The similar phenomenon was also observed for rectangular sheet-like WO<sub>3</sub> crystal with predominant {002} facet as photocatalysts for CO<sub>2</sub> reduction into CH<sub>4</sub>.<sup>[183]</sup> Cd-rich CdSe nanocrystals below a certain critical size are efficient photocatalysts for CO<sub>2</sub> fixation in the presence of light, and bulk CdSe surfaces fail.<sup>[232,233]</sup> Small carbon nanoparticles of sub-10 nm in size harvest visible photons for subsequent charge separation on the particle surface to drive the efficient photocatalytic conversion of CO<sub>2</sub> into formic acid, which represents a highly promising new platform for visible-light photocatalysis of CO<sub>2</sub> conversion.<sup>[234]</sup>

#### 3.1.2. Cation or Anion Doping

The potential of VB of most of currently developed metal oxide photocatalysts is normally located about 3 eV vs NHE,<sup>[235]</sup> which leads to that the metal oxide photocatalysts for CO<sub>2</sub> reduction only utilize the UV light. Replacing partial O in the metal oxides by anion such as N and C is a convenient strategy to raise VB maxima and narrow bandgap of metal oxides for efficiently widen the wavelength range of solar radiation that can be absorbed. N-doped mesoporous TiO<sub>2</sub> with Pt, Au, and Ag as co-catalysts absorbed light up to 500 nm in the visible light spectrum, which contributes to photocatalytic reduction of CO<sub>2</sub> under visible light irradiation ( $\lambda > 420$  nm).<sup>[137]</sup> N-doped Ta<sub>2</sub>O<sub>5</sub> linked with Ru complex electrocatalysts showed selective conversion of CO<sub>2</sub> to HCOOH under visible light irradiation (410 <  $\lambda$  < 750 nm).<sup>[170,171]</sup> N-doped InTaO<sub>4</sub> with Ni@NiO core shell nanostructure as co-catalyst displayed photocatalytic reduction of CO<sub>2</sub> with H<sub>2</sub>O into CH<sub>3</sub>OH under light irradiation (390 <  $\lambda$  < 770 nm).<sup>[176]</sup> I-doped TiO<sub>2</sub> exhibited activity for photocatalytic reduction of CO<sub>2</sub> with H<sub>2</sub>O into CO under visible light irradiation ( $\lambda > 400$  nm).<sup>[133]</sup> The p-n hybrid system of hollow CuO nanocubes decorated with nanoparticles of titanium oxynitride (TiO<sub>2-x</sub>N<sub>x</sub>) photocatalytically converted CO<sub>2</sub> into CH<sub>4</sub> at ambient temperature.<sup>[134]</sup>

Cations such as Cu,<sup>[95,130,140]</sup> Ni,<sup>[128,147]</sup> Ce,<sup>[130,146]</sup> and Ag<sup>[124,135]</sup> are usually doped into the lattice of TiO<sub>2</sub>, creating impurity energy levels within the band gap. The behavior of charge carriers such as the recombination rate and interfacial electron-transfer rate is significantly influenced by the doping of cation.<sup>[236]</sup> Ce-doped TiO<sub>2</sub> enhanced photocatalytic activity for CO<sub>2</sub> reduction than pure TiO<sub>2</sub>, mainly due to that the Ce<sup>3+</sup>/Ce<sup>4+</sup> redox couple facilitates electron-hole separation.<sup>[146]</sup> Ce atoms

**Table 1.** Summary of semiconductor-based systems on photocatalytic reduction of CO<sub>2</sub> (centralized from 1993 to 2013).<sup>a)</sup>

Light source	Catalyst	Co-catalyst	Reaction medium	Products	Efficiency	Reference (year)
UV Hg lamp (254 nm)	TiO <sub>2</sub> nanoparticles (4.5 to 29 nm)	–	CO <sub>2</sub> -bubbled NaOH solution	CH <sub>4</sub> ,* CH <sub>3</sub> OH	–	[82] (2009)
visible light $\lambda > 420$ nm	N-doped mesoporous TiO <sub>2</sub>	Pt, Au, or Ag	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[137] (2012)
solar simulating light (AM1.5)	hybrid CuO-TiO <sub>2-x</sub> N <sub>x</sub> hollow nanocubes	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* C <sub>2</sub> H <sub>6</sub> , C <sub>3</sub> H <sub>8</sub> , C <sub>4</sub> H <sub>10</sub>	–	[137] (2012)
visible light $\lambda > 400$ nm	I-doped TiO <sub>2</sub> nanoparticles	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO	–	[133] (2011)
UV Xe lamp 250 < $\lambda$ < 400 nm	Cu-doped TiO <sub>2</sub> -SiO <sub>2</sub> mesoporous composite particles	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO	–	[95] (2011)
visible light $\lambda > 400$ nm	Cu, I co-doped TiO <sub>2</sub> nanoparticles	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO,* CH <sub>4</sub>	–	[145] (2012)
UV HgXe lamp	Cu, Ce co-doped TiO <sub>2</sub> nanoparticles	–	CO <sub>2</sub> bubbled solution	CH <sub>3</sub> OH	QE = 26.5%	[140] (2012)
UV lamp (365 nm)	Cu, Mn co-doped TiO <sub>2</sub> nanoparticles	–	CO <sub>2</sub> bubbled NaOH solution	CH <sub>3</sub> OH	–	[130] (2011)
UV Xe lamp	Ce-doped TiO <sub>2</sub> /SBA-15	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO,* CH <sub>4</sub>	–	[146] (2012)
UV lamp (254 or 365 nm) or incandescent lamp (400–780 nm)	Ni, N co-doped TiO <sub>2</sub> nanoparticles	–	CO <sub>2</sub> -bubbled NaOH solution containing Na <sub>2</sub> SO <sub>3</sub>	CH <sub>3</sub> OH	–	[128] (2011)
Hg lamp (254 nm)	Ag-doped TiO <sub>2</sub> nanoparticles	–	CO <sub>2</sub> bubbled solution	CH <sub>4</sub> ,* CH <sub>3</sub> OH	–	[124] (2010)
UV Hg lamp	rutile TiO <sub>2</sub> nanoparticle modified anatase TiO <sub>2</sub> nanorods	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* CH <sub>3</sub> OH, H <sub>2</sub> , CO	–	[105] (2012)
visible light $\lambda > 400$ nm	TiO <sub>2</sub> anatase–brookite junction	–	NaHCO <sub>3</sub> solution	CH <sub>3</sub> OH	QE = 0.0717%	[141] (2012)
visible light $\lambda > 400$ nm	CdS(or Bi <sub>2</sub> S <sub>3</sub> )/TiO <sub>2</sub> nanotubes	–	CO <sub>2</sub> -bubbled NaOH solution containing Na <sub>2</sub> SO <sub>3</sub>	CH <sub>3</sub> OH	–	[136] (2012)
visible light $\lambda > 420$ nm	AgBr/TiO <sub>2</sub> nanocomposites	–	CO <sub>2</sub> -saturated KHCO <sub>3</sub> solution containing NaOH	CH <sub>4</sub> ,* CH <sub>3</sub> OH, C <sub>2</sub> H <sub>5</sub> OH, CO	–	[127] (2011)
concentrated natural sunlight	Fe and Cu co-doped TiO <sub>2</sub> sensitized with Ru dye	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[121] (2008)
visible light $\lambda > 420$ nm	enzyme-modified TiO <sub>2</sub> nanoparticles	–	CO <sub>2</sub> bubbled solution Containing MES	CO	–	[126] (2010)
visible light $\lambda > 420$ nm	copper(II) dye-sensitized TiO <sub>2</sub> nanoparticles	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[144] (2012)
tungsten–halogen lamp.	Cobalt-phthalocyanine (CoPc) dye/TiO <sub>2</sub>	–	CO <sub>2</sub> -bubbled NaOH solution containing Na <sub>2</sub> SO <sub>3</sub>	HCOOH,* CO, HCHO, CH <sub>3</sub> OH, CH <sub>4</sub>	–	[118] (2007)
solar simulating light (AM1.5)	N-doped TiO <sub>2</sub> nanotube arrays	Cu and Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* C <sub>2</sub> H <sub>4</sub> , C <sub>2</sub> H <sub>6</sub>	–	[83] (2009)
UV Xe lamp	TiO <sub>2</sub> rods with {010} facets	1 wt% Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[91] (2011)
UV Xe lamp	1D TiO <sub>2</sub> film	Pt (1.04 nm)	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* CO	QE = 2.41%	[106] (2012)
solar simulating light (AM1.5)	TiO <sub>2</sub> nanotube arrays	Cu <sub>0.33</sub> -Pt <sub>0.67</sub>	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* C <sub>2</sub> H <sub>4</sub> , C <sub>2</sub> H <sub>6</sub>	–	[107] (2012)
UV Hg lamp	TS-1 zeolites	–	CO <sub>2</sub> and H <sub>2</sub>	CH <sub>4</sub>	–	[42] (1995)
UV Hg lamp	ex-Ti-oxide/Y-zeolite	1 wt% Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* CH <sub>3</sub> OH	–	[44] (1997)
UV Hg lamp	Ti-MCM-41 or Ti-MCM-48 zeolites	1 wt% Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* CH <sub>3</sub> OH	–	[48] (1998)
UV Hg lamp	Ti- $\beta$ (F) or Ti- $\beta$ (OH) zeolites	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* CH <sub>3</sub> OH	–	[56] (2001)
UV Hg lamp	Titanium porous silica thin films	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* CH <sub>3</sub> OH	QE = 0.28%	[60] (2002)
UV Hg lamp	Ti-SBA-15	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* CH <sub>3</sub> OH	–	[70] (2005)

**Table 1.** Continued.

Light source	Catalyst	Co-catalyst	Reaction medium	Products	Efficiency	Reference (year)
UV lamp	Ti-KIT-6	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[109] (2013)
UV Xe lamp	porous silica supported TiO <sub>2</sub>	0.5 wt% Cu	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO,* CH <sub>4</sub>	QE = 1.41%	[86] (2010)
UV Xe lamp	Metal complex doped UiO-67 MOF	–	CO <sub>2</sub> -saturated MeCN solution	CO,* H <sub>2</sub>	–	[246] (2011)
visible light 420 < $\lambda$ < 800 nm	NH <sub>2</sub> -MIL-125(Ti) MOF	–	CO <sub>2</sub> -saturated MeCN/TEOA (5:1) solution	HCOOH	–	[247] (2012)
visible light 420 < $\lambda$ < 800 nm	NH <sub>2</sub> -Uio-66(Zr) MOF	–	CO <sub>2</sub> -saturated MeCN/TEOA (5:1) solution	HCOOH	–	[248] (2013)
UV Xe lamp	TiO <sub>2</sub> /ZnO mesoporous “french fries”	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[96] (2011)
UV Xe lamp	hollow anatase TiO <sub>2</sub> single crystals with {101} facets	1 wt% RuO <sub>2</sub>	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[100] (2012)
UV Xe lamp	TiO <sub>2</sub> ultrathin nanosheets with 95% of exposed {100} facet	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[113] (2013)
UV Xe lamp	P25	Pt@Cu <sub>2</sub> O	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* CO, H <sub>2</sub>	–	[114] (2013)
UV Hg lamp	TiO <sub>2</sub> nanoparticles	Pt or Pd	CO <sub>2</sub> -bubbled solution	CH <sub>4</sub> ,* C <sub>2</sub> H <sub>6</sub>	–	[36] (1993)
UV Xe lamp	TiO <sub>2</sub> nanoparticles	Pd and RuO <sub>2</sub>	CO <sub>2</sub> -bubbled NaOH solution containing Na <sub>2</sub> SO <sub>3</sub>	HCOOH	–	[59] (2001)
UV Xe lamp	nafion coated Pd–TiO <sub>2</sub> particles	1.0 wt% Pd	CO <sub>2</sub> -saturated Na <sub>2</sub> CO <sub>3</sub> solution	CH <sub>4</sub> ,* C <sub>2</sub> H <sub>6</sub>	–	[101] (2012)
UV Hg lamp	TiO <sub>2</sub> nanoparticles	Cu	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>3</sub> OH,* CH <sub>4</sub>	–	[39] (1994)
UV lamp	TiO <sub>2</sub> nanoparticles	3.0 wt% CuO	CO <sub>2</sub> -bubbled KHCO <sub>3</sub> aqueous solution	CH <sub>3</sub> OH	QE = 19.23%	[116] (2005)
UV Hg lamp (254 nm)	TiO <sub>2</sub> nanoparticles	2.0 wt% Cu	CO <sub>2</sub> -bubbled NaOH solution	CH <sub>3</sub> OH, O <sub>2</sub>	QE = 10%, $\eta$ = 2.5%	[62] (2002)
UV Hg lamp (365 nm)	TiO <sub>2</sub> coated on optical fiber	1.2 wt% Cu	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>3</sub> OH	–	[73] (2005)
UV lamp	TiO <sub>2</sub> nanoparticles	0.03 wt% Cu	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[102] (2012)
UV Xe lamp	mesoporous TiO <sub>2</sub> /graphitic carbon microspheres	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[115] (2013)
UV lamp (365 nm)	MWCNTs supported TiO <sub>2</sub>	–	CO <sub>2</sub> -saturated H <sub>2</sub> O solution	HCOOH,* CH <sub>4</sub> , C <sub>2</sub> H <sub>5</sub> OH	–	[77] (2007)
visible light $\lambda$ > 400 nm	CNTs@Ni-doped TiO <sub>2</sub> nanocomposites	Ni	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[147] (2013)
UV Xe lamp	graphene-Ti <sub>0.91</sub> O <sub>2</sub> hollow spheres	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO,* CH <sub>4</sub>	–	[39] (2012)
UV Xe lamp	graphene-TiO <sub>2</sub> nanosheets	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	C <sub>2</sub> H <sub>6</sub> , CH <sub>4</sub>	–	[111] (2013)
visible light 400 < $\lambda$ < 850 nm	less defective graphene-P25 nanocomposites	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[90] (2011)
visible light $\lambda$ > 400 nm	self-doped SrTiO <sub>3-<math>\delta</math></sub> particles	0.3%wt Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	E = 0.21%	[151] (2011)
UV Hg lamp	SrTiO <sub>3</sub> nanocrystals	7 wt% Ag	CO <sub>2</sub> bubbled CH <sub>3</sub> OH solution	HCOOH	–	[152] (2012)
UV Xe lamp	leaf-architected SrTiO <sub>3</sub>	1 wt% Au	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO,* CH <sub>4</sub>	–	[153] (2013)
UV Hg or Xe lamp	K <sub>2</sub> Ti <sub>6</sub> O <sub>13</sub> particles	Pt	CO <sub>2</sub> -saturated H <sub>2</sub> O solution	H <sub>2</sub> ,* CH <sub>4</sub> , HCOOH, HCHO	–	[155] (2003)
UV Hg lamp	BaLa <sub>4</sub> Ti <sub>4</sub> O <sub>15</sub> nanoplates	Ag	CO <sub>2</sub> bubbled H <sub>2</sub> O solution	CO,* HCOOH, H <sub>2</sub> , O <sub>2</sub>	–	[157] (2011)
UV Hg lamp	ZrO <sub>2</sub>	1 wt% Cu	NaHCO <sub>3</sub> solution	CO	–	[158] (1993)
UV Xe lamp	SiO <sub>2</sub> -Pillared HNb <sub>3</sub> O <sub>8</sub>	0.4 wt% Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[162] (2012)
UV Xe lamp	HNb <sub>3</sub> O <sub>8</sub> nanobelts	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[163] (2012)

Table 1. Continued.

Light source	Catalyst	Co-catalyst	Reaction medium	Products	Efficiency	Reference (year)
visible light $500 < \lambda < 900$ nm	InNbO <sub>4</sub> (SSR)	1 wt% NiO	KHCO <sub>3</sub> aqueous solution	CH <sub>3</sub> OH	–	[164] (2012)
UV Hg lamp	LiNbO <sub>3</sub> particles	–	CO <sub>2</sub> -saturated H <sub>2</sub> O solution	HCOOH,* HCHO	QE = 2%	[166] (2011)
UV Xe lamp	NaNbO <sub>3</sub> nanowires	Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[165] (2011)
UV Xe lamp	cubic and orthorhombic NaNbO <sub>3</sub> nanoparticles	Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[167] (2012)
UV Xe lamp	ANbO <sub>3</sub> (A = Na, K) (SSR)	Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[168] (2012)
visible light $410 < \lambda < 750$ nm	N- DopedTa <sub>2</sub> O <sub>5</sub>	Ru-complex	CO <sub>2</sub> -saturated MeCN/TEOA (5:1) solution	HCOOH,* H <sub>2</sub> , CO	QE = 1.9%	[170] (2010)
visible light $410 < \lambda < 750$ nm	N- DopedTa <sub>2</sub> O <sub>5</sub>	Ru-complex	CO <sub>2</sub> -saturated MeCN/TEOA (5:1) solution	HCOOH,* CO	–	[171] (2011)
halogen lamp	InTaO <sub>4</sub> (SSR)	1 wt% NiO	KHCO <sub>3</sub> aqueous solution	CH <sub>3</sub> OH	QE = 2.45%, $\eta = 1.158\%$	[172] (2007)
halogen lamp $400 < \lambda < 1100$ nm	InTaO <sub>4</sub> particles	1 wt% NiO	CO <sub>2</sub> bubbled NaOH solution	CH <sub>3</sub> OH	QE = 0.063%	[174] (2010)
Xe lamp $390 < \lambda < 770$ nm	N-doped InTaO <sub>4</sub> (SSR)	Ni@NiO core shell	CO <sub>2</sub> bubbled H <sub>2</sub> O solution	CH <sub>3</sub> OH	–	[176] (2011)
UV Hg–Xe lamp	ATaO <sub>3</sub> (A = Li, Na, K) (SSR)	–	CO <sub>2</sub> and H <sub>2</sub>	CO	–	[177] (2010)
visible light $\lambda > 400$ nm	BiVO <sub>4</sub> monoclinic crystals	–	CO <sub>2</sub> bubbled H <sub>2</sub> O solution	C <sub>2</sub> H <sub>5</sub> OH,* CH <sub>3</sub> OH	–	[178] (2009)
visible light $\lambda > 420$ nm	Fe <sub>2</sub> V <sub>4</sub> O <sub>13</sub> nanoribbons	0.5 wt% Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[179] (2013)
visible light $\lambda > 420$ nm	Na <sub>2</sub> V <sub>6</sub> O <sub>16</sub> nanoribbons	Pt and RuO <sub>2</sub>	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[180] (2013)
UV Xe lamp	Bi <sub>6</sub> Mo <sub>2</sub> O <sub>15</sub> sub-microwires	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> , O <sub>2</sub>	–	[181] (2013)
visible light $\lambda > 420$ nm	WO <sub>3</sub> nanosheets	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[182] (2012)
UV Xe lamp	rectangular sheet-like WO <sub>3</sub>	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[183] (2012)
visible light $\lambda > 420$ nm	W <sub>18</sub> O <sub>49</sub> nanowires	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[184] (2012)
visible light $\lambda > 420$ nm	Bi <sub>2</sub> WO <sub>6</sub> nanoplates	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[185] (2012)
visible light $\lambda > 420$ nm	Bi <sub>2</sub> WO <sub>6</sub> hollow microspheres	–	CO <sub>2</sub> bubbled H <sub>2</sub> O solution	CH <sub>3</sub> OH	–	[186] (2012)
UV Xe lamp	mesoporous In(OH) <sub>3</sub>	0.5 wt% Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[187] (2013)
UV Xe lamp	porous $\beta$ -Ga <sub>2</sub> O <sub>3</sub>	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO,* CH <sub>4</sub>	QE = 3.993%	[189] (2012)
UV Xe lamp	mesoporous ZnGa <sub>2</sub> O <sub>4</sub>	1 wt% RuO <sub>2</sub>	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[191] (2010)
UV Xe lamp	ZnGa <sub>2</sub> O <sub>4</sub> nanocubes	0.5 wt% RuO <sub>2</sub>	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> , O <sub>2</sub>	–	[192] (2012)
visible light $\lambda > 420$ nm	ZnAl <sub>2</sub> O <sub>4</sub> -modified mesoporous ZnGaNO	0.5 wt% Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[202] (2012)
UV Xe lamp	CuGaO <sub>2</sub> (SSR)	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO,* CH <sub>4</sub>	–	[193] (2012)
UV Xe lamp	Zn <sub>2</sub> GeO <sub>4</sub> nanobelts	Pt, RuO <sub>2</sub>	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[194] (2010)
UV Xe lamp	Zn <sub>2</sub> GeO <sub>4</sub> nanorods	3 wt% RuO <sub>2</sub>	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO,* CH <sub>4</sub>	–	[195] (2011)
UV Xe lamp	mesoporous Zn <sub>2</sub> GeO <sub>4</sub>	1 wt% Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	QE = 0.2%	[196] (2011)
UV Xe lamp	ZIF-8/Zn <sub>2</sub> GeO <sub>4</sub> nanorods	1 wt% Pt	CO <sub>2</sub> bubbled Na <sub>2</sub> SO <sub>3</sub> solution	CH <sub>3</sub> OH	–	[197] (2013)
visible light $\lambda > 420$ nm	sheaf-like, hyperbranched Zn <sub>1.7</sub> GeN <sub>1.8</sub> O	Pt and RuO <sub>2</sub>	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	QE = 0.024%	[200] (2012)
visible light $\lambda > 400$ nm	mesoporous (Zn <sub>1+x</sub> Ge)(N <sub>2</sub> O <sub>x</sub> )	1 wt% Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> ,* CO	–	[201] (2012)
UV Xe lamp	4.5(ZnGa <sub>2</sub> O <sub>4</sub> ):(Zn <sub>2</sub> GeO <sub>4</sub> ) solid solution	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub> , O <sub>2</sub>	–	[237] (2013)

**Table 1.** Continued.

Light source	Catalyst	Co-catalyst	Reaction medium	Products	Efficiency	Reference (year)
UV Xe lamp	In <sub>2</sub> Ge <sub>2</sub> O <sub>7</sub> (En) nanowires	1 wt% Pt	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO, O <sub>2</sub>	–	[198] (2012)
UV Xe lamp	Zn <sub>2</sub> SnO <sub>4</sub> nanoplates	Pt and RuO <sub>2</sub>	CO <sub>2</sub> and H <sub>2</sub> O vapor	CH <sub>4</sub>	–	[199] (2012)
UV Xe lamp	ZnS microcrystals	Cd	CO <sub>2</sub> bubbled 2-propanol solution	HCOOH,* H <sub>2</sub>	QE = 32.5%	[205] (1995)
UV ( $\lambda > 290$ nm)	ZnS nanoparticles supported on silica	–	CO <sub>2</sub> bubbled solution 2,5- containing DHF	HCOOH	–	[207] (1997)
UV ( $\lambda > 290$ nm)	ZnS nanocrystals	–	CO <sub>2</sub> bubbled solution containing triethylamine	HCOOH,* CO, H <sub>2</sub>	–	[209] (1998)
visible light $\lambda > 400$ nm	CdS nanocrystals	–	CO <sub>2</sub> bubbled solution containing triethylamine	CO	–	[206] (1997)
UV Hg-lamp	CdS nanocrystals	–	CO <sub>2</sub> bubbled solution	HCOOH,* HCHO	–	[208] (1998)
UV Hg-lamp	CdS particles	–	CO <sub>2</sub> bubbled 2-propanol solution	HCOOH,* CO, H <sub>2</sub>	–	[210] (1998)
UV ( $\lambda > 290$ nm)	CdS nanoparticles supported on silica	–	NaHCO <sub>3</sub> solution containing Na <sub>2</sub> SO <sub>3</sub>	HCOOH,* HCHO, oxalic acid	–	[211] (2002)
UV Hg-lamp (254 nm)	CdS nanoparticles deposited on montmorillonite	–	CO <sub>2</sub> -saturated NaOH solutions	H <sub>2</sub> ,* CH <sub>4</sub> , CO	–	[212] (2011)
visible light $\lambda > 420$ nm	enzyme coupled CdS nanocrystals	–	CO <sub>2</sub> -saturated MES solution	CO	–	[213] (2012)
visible light $\lambda > 400$ nm	Bi <sub>2</sub> S <sub>3</sub> /CdS	–	CO <sub>2</sub> -bubbled NaOH solution containing Na <sub>2</sub> SO <sub>3</sub>	CH <sub>3</sub> OH	–	[214] (2011)
UV Hg lamp	MnS particles	–	NaHCO <sub>3</sub> solution containing NaHS	HCOOH,* CO	QE = 4.2%	[215] (2004)
visible light $\lambda > 400$ nm	Cu <sub>x</sub> Ag <sub>y</sub> In <sub>z</sub> Zn <sub>k</sub> S solid solutions	RuO <sub>2</sub> or Rh <sub>1.32</sub> Cr <sub>0.66</sub> O <sub>3</sub>	NaHCO <sub>3</sub> solution	CH <sub>3</sub> OH	–	[216] (2011)
visible light 400 < $\lambda$ < 800 nm	Cu <sub>2</sub> ZnSnS <sub>4</sub> photocathode	Ru-complex	CO <sub>2</sub> -saturated solution	HCOOH	–	[217] (2011)
661 nm	p-Si photocathode	Re-complex	PEC cell	CO	$\eta = 9.3\%$	[218] (2010)
solar simulating light (AM1.5)	p-InP photocathode	Ru-complex	NaHCO <sub>3</sub> solution (PEC cell)	HCOOH,* O <sub>2</sub>	$\eta = 0.04\%$	[222] (2011)
solar simulating light (AM1.5)	p-InP photocathode	Ru-complex	NaHCO <sub>3</sub> solution (PEC cell)	HCOOH,* O <sub>2</sub>	$\eta = 0.14\%$	[223] (2013)
UV 365 nm	p-GaP photocathode	–	PEC cell	CH <sub>3</sub> OH	$\eta = 9.3\%$ , QE = 44% at –0.5V	[224] (2008)
visible light 425 < $\lambda$ < 720 nm	carbon nanoparticles of sub-10 nm	Au	CO <sub>2</sub> -saturated solution	HCOOH	–	[234] (2011)
UV Xe lamp	layered Zn-Al LDH	–	CO <sub>2</sub> and H <sub>2</sub>	CO,* CH <sub>3</sub> OH	–	[258] (2011)
UV Hg-Xe lamp	layered Mg-In LDH	–	CO <sub>2</sub> -saturated H <sub>2</sub> O solution	CO,* O <sub>2</sub>	–	[259] (2012)
UV Xe lamp	layered Zn–Cu–Ga LDH	–	CO <sub>2</sub> and H <sub>2</sub>	CH <sub>3</sub> OH,* CO	–	[260] (2012)
visible light $\lambda > 420$ nm	g-C <sub>3</sub> N <sub>4</sub>	–	CO <sub>2</sub> and H <sub>2</sub> O vapor	CO	–	[262] (2012)
visible light $\lambda > 420$ nm	mesoporous g-C <sub>3</sub> N <sub>4</sub> flakes	–	CO <sub>2</sub> -saturated NaOH solution	CH <sub>3</sub> OH,* C <sub>2</sub> H <sub>5</sub> OH	QE = 0.18%	[263] (2013)
visible light $\lambda > 400$ nm	TaON–RuBLRu'	Ag	CO <sub>2</sub> bubbled MeOH solution	HCOOH,* H <sub>2</sub> , CO	$\Delta G^\circ = +83.0$ kJ/mol	[267] (2013)

<sup>a)</sup>The symbol '\*' refers to the major product. 'SSR' refers to samples prepared by solid-state reaction. 'QE' refers to apparent quantum yield. ' $\eta$ ' refers to solar energy conversion efficiency. 'MWCNTs' and 'CNTs' refer to multiwalled carbon nanotubes and carbon nanotubes, respectively.



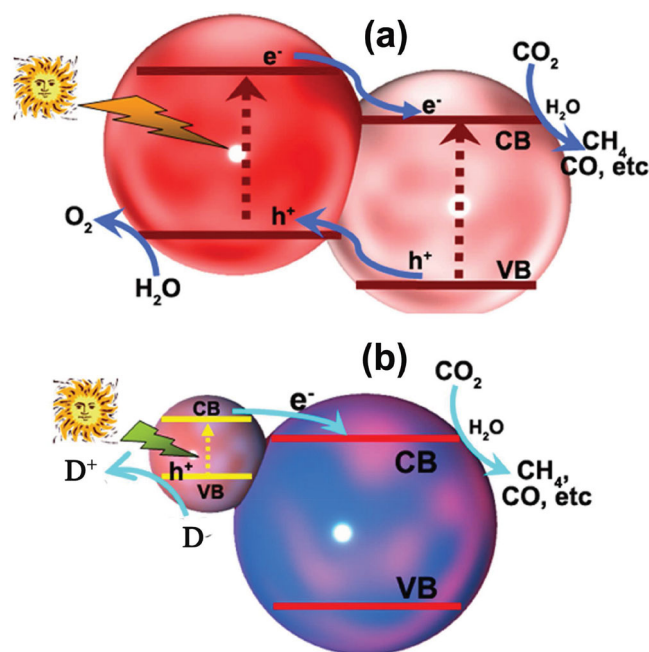
enter the crystal lattice of  $\text{TiO}_2$  to bring lattice distortion and activate adsorbed  $\text{H}_2\text{O}$  and  $\text{CO}_2$  molecules. Nevertheless, excessively high Ce concentration destroyed photocatalytic activity for  $\text{CO}_2$  reduction as the formation of recombination centers by excess Ce accelerated electron–hole recombination, which was also observed in Cu-doped  $\text{TiO}_2$ .<sup>[95]</sup> Ce and Cu co-doped  $\text{TiO}_2$  improved the efficiency for reduction of  $\text{CO}_2$  into  $\text{CH}_3\text{OH}$ .<sup>[130]</sup> Ce doping affected the reaction more profoundly than that of Cu doping. Ag-doped  $\text{TiO}_2$  (1–7 wt% Ag) caused the yield of  $\text{CH}_4$  and  $\text{CH}_3\text{OH}$  by two mechanisms: up to 5wt% Ag doping in  $\text{TiO}_2$  resulted in the shift of absorption edge towards visible light and increased generation of electron–hole pairs, and with Ag content above 5 wt%, the formation of Schottky barrier at the metal–semiconductor interface around Ag metallic clusters enhanced separation of electrons and holes.<sup>[124]</sup>

### 3.1.3. Solid Solutions

The solid solutions exhibit better performance than the single components comprising the solid solution. Solid solutions such as  $(\text{Ga}_{1-x}\text{Zn}_x)(\text{N}_{1-x}\text{O}_x)$ ,<sup>[202]</sup>  $(\text{Zn}_{1+x}\text{Ge})(\text{N}_2\text{O}_x)$ <sup>[200,201]</sup> and zinc gallogermanate solid solution<sup>[237]</sup> was utilized for photocatalytic reduction of  $\text{CO}_2$ . Yellow  $\text{Zn}_{1.7}\text{GeN}_{1.8}\text{O}$  solid solution with a bandgap of 2.6 eV exhibited highest activity for  $\text{CO}_2$  reduction with apparent quantum yield of 0.024% at wavelength of  $420 \pm 15$  nm when loading 1 wt% Pt and 1 wt%  $\text{RuO}_2$  co-catalysts, originated from p–d repulsion of N2p and Zn3d orbitals which lifts the top of VB and then narrows the band gap.<sup>[200]</sup> Zinc gallogermanate (denoted as  $(4.5(\text{ZnGa}_2\text{O}_4):(\text{Zn}_2\text{GeO}_4))$  solid solution that incorporates  $\text{Zn}_2\text{GeO}_4$  into  $\text{ZnGa}_2\text{O}_4$  can narrow band gap by the upshift of VB edge from the increased p–d ( $\text{O}2\text{p}$ – $\text{Zn}3\text{d}$ ) repulsion effect through incorporation of s and p orbitals of Ge, and the downshift of CB edge by introducing the low-energy orbital of Ge.<sup>[237]</sup> The zinc gallogermanate solid solution possesses a light-hole effective mass which is beneficial for improving hole mobility, and thus enhances the ability for water oxidation to provide protons for  $\text{CO}_2$  reduction.  $\text{Cu}_x\text{Ag}_y\text{In}_z\text{Zn}_k\text{S}$  solid solutions (bandgap < 2 eV) customized with  $\text{RuO}_2$  or  $\text{Rh}_{1.32}\text{Cr}_{0.66}\text{O}_3$  co-catalyst showed high photocatalytic activity for reduction of  $\text{CO}_2$  into  $\text{CH}_3\text{OH}$  under visible light irradiation.<sup>[216]</sup>

### 3.1.4. Heterostructures

Heterostructures consisting of two or more materials can optimize charge separation and photocatalytic properties by providing an additional control of the electron and hole wave functions in these materials (i.e., wave function engineering).<sup>[238]</sup> The heterostructures (type II) can promote the spatial separation of electron–hole pairs by transferring photoexcited electrons in the higher CB to the lower CB and holes in the lower VB to the higher VB (Figure 4a).  $\text{Cu}_2\text{O}/\text{SiC}$ ,<sup>[239]</sup>  $\text{Bi}_2\text{S}_3/\text{CdS}$ ,<sup>[214]</sup> and  $\text{TiO}_2/\text{ZnO}$ <sup>[96]</sup> heterostructures have been constructed for photocatalytic reduction of  $\text{CO}_2$ . Rutile  $\text{TiO}_2$  nanoparticle modified anatase  $\text{TiO}_2$  nanorods with exposed {010} facets showed higher photocatalytic reduction of  $\text{CO}_2$  into  $\text{CH}_4$  than pure anatase  $\text{TiO}_2$  nanorods, due to the transfer of



**Figure 4.** Schematic illustrations of: a) semiconductor heterostructures, b) sensitization of a wide band gap semiconductor with narrow-bandgap semiconductors or dye molecules for photocatalytic reduction of  $\text{CO}_2$  with  $\text{H}_2\text{O}$  (D is a hole scavenger).

electrons from anatase to rutile under UV light irradiation.<sup>[105]</sup> Moreover,  $\text{TiO}_2$  anatase–brookite junction exhibited remarkable photocatalytic performance for reduction of  $\text{CO}_2$  into  $\text{CH}_3\text{OH}$ , resulting from the transfer of electrons from brookite to anatase.<sup>[141]</sup>

### 3.1.5. Sensitization

Sensitizing wide-bandgap semiconductors with narrow-bandgap semiconductors or dye molecules is an effective strategy to realize visible light reduction of  $\text{CO}_2$  through the transfer of photogenerated electrons from narrow-bandgap semiconductors or dye molecules to the CB of wide-bandgap semiconductors (Figure 4b). Narrow-bandgap semiconductors such as  $\text{CdS}$ ,<sup>[136]</sup>  $\text{Bi}_2\text{S}_3$ ,<sup>[136]</sup>  $\text{CdSe}$  quantum dots (QDs),<sup>[125]</sup>  $\text{PbS}$  QDs,<sup>[131]</sup> and  $\text{AgBr}$ <sup>[127]</sup> have been coupled with  $\text{TiO}_2$  for photocatalytic reduction of  $\text{CO}_2$  under visible light irradiation.  $\text{CdS}$  or  $\text{Bi}_2\text{S}_3$  sensitized  $\text{TiO}_2$  nanotubes showed photocatalytic reduction of  $\text{CO}_2$  into  $\text{CH}_3\text{OH}$  using  $\text{Na}_2\text{SO}_3$  as a hole scavenger under visible light irradiation.<sup>[136]</sup> Photogenerated holes of  $\text{AgBr}/\text{TiO}_2$  photocatalyst react with adsorbed  $\text{H}_2\text{O}$  molecules on the catalyst surface to form  $\cdot\text{OH}$  radicals and  $\text{H}^+$  which take part in the formation of  $\text{CH}_4$  and  $\text{CH}_3\text{OH}$  under visible light irradiation ( $\lambda > 420$  nm).<sup>[127]</sup> The yields of  $\text{CO}$ ,  $\text{CH}_4$ , and  $\text{C}_2\text{H}_6$  over  $\text{PbS}$  QDs (3–5 nm) sensitized  $\text{TiO}_2$  with Cu co-catalyst were also detected.<sup>[131]</sup>

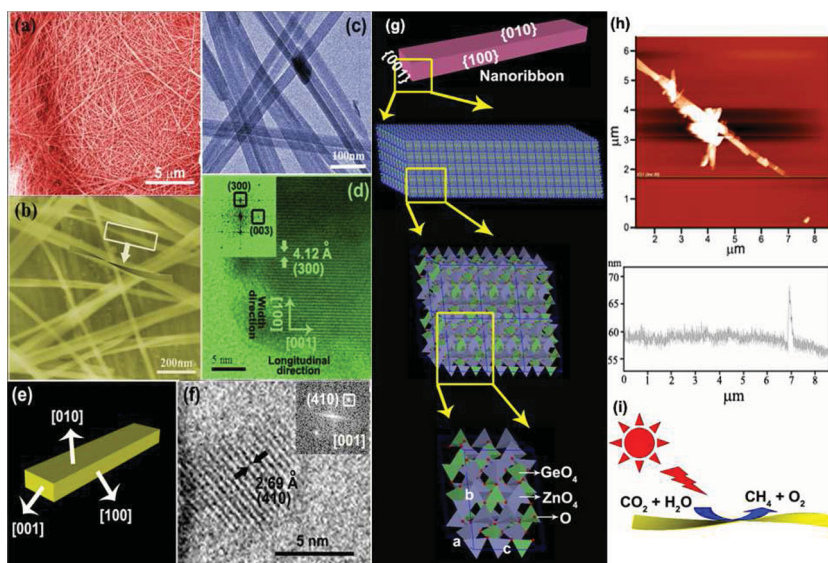
Dye molecules have been linked with  $\text{TiO}_2$  for photocatalytic reduction of  $\text{CO}_2$  under visible light irradiation.<sup>[118–123,126,132,143,144,148]</sup> Compared with totally inactive pure  $\text{TiO}_2$  under visible-light illumination,  $[\text{Ru}(\text{Bpy})_3]^{2+}$

dye-sensitized  $\text{TiO}_2$  films showed efficient activity for reduction of  $\text{CO}_2$  into  $\text{CH}_4$ , originating from transfer of the catalytically active electrons from organic dye to  $\text{TiO}_2$ .<sup>[119,120]</sup> Fe and Cu co-doped  $\text{TiO}_2$  sensitized with Ru dye (N3 dye) were employed to reduce  $\text{CO}_2$  with  $\text{H}_2\text{O}$  to  $\text{CH}_4$  in an optical-fiber photo-reactor under concentrated natural sunlight.<sup>[121]</sup>  $\text{TiO}_2$  functionalized with the carbon monoxide dehydrogenases (CODHs) enzyme and a ruthenium bipyridyl sensitizer (RuP) was constructed for reduction of  $\text{CO}_2$  into CO under visible light irradiation.<sup>[126,132]</sup> RuP injects electrons into the CB of  $\text{TiO}_2$ , and these electrons are transferred into CODH and the active sites where  $\text{CO}_2$  is reduced to CO.  $\text{TiO}_2$  with an air-stable copper (I) dye as a sensitizer exhibited efficient light harvesting and high efficiency for reduction of  $\text{CO}_2$  into  $\text{CH}_4$  under visible light irradiation ( $\lambda > 420 \text{ nm}$ ).<sup>[144]</sup> Cobalt-phthalocyanine (CoPc) dye<sup>[118,122,123]</sup> or zinc-phthalocyanine (ZnPc) dye<sup>[143]</sup> has been adsorbed on  $\text{TiO}_2$  for  $\text{CO}_2$  reduction under ambient conditions. CoPc- $\text{TiO}_2$  showed high selectivity for the formation of  $\text{HCOOH}$  while a small quantity of CO,  $\text{HCHO}$ ,  $\text{CH}_3\text{OH}$  and  $\text{CH}_4$  was generated in NaOH solution containing  $\text{Na}_2\text{SO}_3$  as a hole scavenger.<sup>[118]</sup>

### 3.2. Nanostructuralization

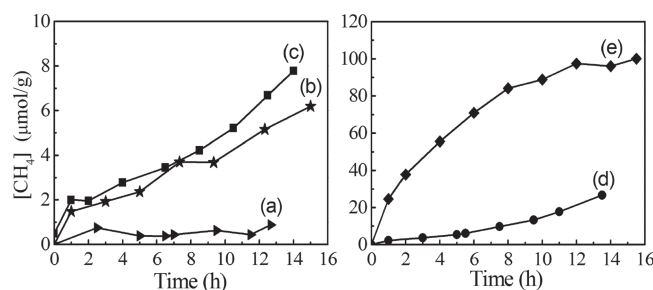
Nanostructures in forms of nanorods, nanowires, nanotubes, nanobelts, etc have been extensively studied for photocatalytic application.<sup>[240,241]</sup> The nanostructured semiconductors are usually of single crystalline phases that exclude the possibility of any grain boundaries and defects in the materials (usually act as recombination sites in polycrystalline materials), which decreases the electron-hole recombination rate. Nanostructures, especially One-dimensional nanostructures, also provide a direct pathway for electron transport, which is favorable for improving the separation of electron-hole pairs.<sup>[242]</sup> Nanostructured semiconductors have proved to exhibit superior performance for photocatalytic reduction of  $\text{CO}_2$  in recent years, such as  $\text{TiO}_2$ ,<sup>[83,88,91,103,105–107,136]</sup>  $\text{Fe}_2\text{V}_4\text{O}_{13}$  nanoribbons,<sup>[179]</sup>  $\text{Na}_2\text{V}_6\text{O}_{16}$  nanoribbons,<sup>[180]</sup>  $\text{HNb}_3\text{O}_8$  nanoribbons,<sup>[163]</sup>  $\text{NaNbO}_3$  nanowires,<sup>[165]</sup>  $\text{In}_2\text{Ge}_2\text{O}_7$  (En) (En = ethylenediamine) hybrid sub-nanowires,<sup>[198]</sup>  $\text{Zn}_2\text{GeO}_4$  nanoribbons and nanorods.<sup>[194,195,197]</sup>

$\text{Zn}_2\text{GeO}_4$  nanoribbons (ca. 7 nm thickness, corresponding to five repeating cell unit) with ultralong and ultrathin geometry greatly improved photocatalytic activity toward reduction of  $\text{CO}_2$  with  $\text{H}_2\text{O}$  into  $\text{CH}_4$  relative to bulk  $\text{Zn}_2\text{GeO}_4$  (Figure 5 and 6).<sup>[194]</sup> The ultrathin geometry of the nanoribbons allows charge carriers to move rapidly from the interior to the surface to participate in the photoreduction reaction. The single crystalline hexagonal prism  $\text{Zn}_2\text{GeO}_4$  nanorods (400 nm length and 50 nm width) also exhibited higher activity for



**Figure 5.** a,b) FE-SEM images, c) TEM and d) HRTEM images of  $\text{Zn}_2\text{GeO}_4$  nanoribbon (the inset of (d) is the FFT pattern obtained from the HRTEM image); e) structural model of the nanoribbon, f) HRTEM image taken along [001], g) corresponding structural model of  $\text{Zn}_2\text{GeO}_4$  nanoribbon, h) AFM image of  $\text{Zn}_2\text{GeO}_4$  nanoribbon and corresponding high profile, and i) schematic illustration of photoreduction of  $\text{CO}_2$  over the nanoribbon. Reproduced with permission.<sup>[194]</sup> Copyright 2010, American Chemical Society.

photocatalytic reduction of  $\text{CO}_2$  into  $\text{CH}_4$ .<sup>[195]</sup>  $\text{In}_2\text{Ge}_2\text{O}_7$  (En) ultrathin nanowire (2–3 nm in diameter) proved to possess photocatalytic activity towards the conversion of  $\text{CO}_2$  to CO and  $\text{O}_2$  in the present water vapor under Xe arc lamp irradiation.<sup>[198]</sup>  $\text{NaNbO}_3$  nanowires with a uniform diameter of about 100 nm and length of up to several tens micrometers exhibited about 20-time significant improvement for photocatalytic reduction of  $\text{CO}_2$  into  $\text{CH}_4$  under UV-vis light irradiation, relative to bulk  $\text{NaNbO}_3$ , due to superior crystallinity, the high surface area, and highly mobile charge carriers.<sup>[165]</sup> 1D  $\text{Fe}_2\text{V}_4\text{O}_{13}$  nanoribbon arrays with a bandgap of 1.83 eV directly grown on a stainless-steel mesh served as a new candidate for reduction of  $\text{CO}_2$  into  $\text{CH}_4$  under visible light irradiation.<sup>[179]</sup> This substrate-supported  $\text{Fe}_2\text{V}_4\text{O}_{13}$  nanoribbons avoids the complex recycling process of powders from solution when photocatalytic reduction of  $\text{CO}_2$  is carried in solution.



**Figure 6.**  $\text{CH}_4$  generation over  $\text{Zn}_2\text{GeO}_4$ : solid-state reaction (a), nanoribbon (b), 1 wt% Pt-loaded nanoribbon (c), 1 wt%  $\text{RuO}_2$ -loaded nanoribbon (d), and 1 wt%  $\text{RuO}_2$  and 1 wt% Pt co-loaded nanoribbons (e) as a function of light irradiation times. Reproduced with permission.<sup>[194]</sup> Copyright 2010, American Chemical Society.

### 3.3. Surface Oxygen Vacancy Engineering

Oxygen vacancy is an intrinsic defect in metal oxides, which is the most reactive site on the surface and able to modify the structure of the materials and change the electronic and chemical properties of the surface.<sup>[243]</sup> The surface oxygen vacancies sites of semiconductors play an important role in CO<sub>2</sub> activation.<sup>[10,31,151,184,233,244,245]</sup> CO<sub>2</sub> molecules prefer to adsorb at oxygen vacancy sites with one oxygen atoms of the CO<sub>2</sub> located at bridging oxygen vacancy defects.<sup>[28,29]</sup> The attraction between an oxygen vacancy and a CO<sub>2</sub> molecule may lower the reactive barrier due to the generation of an unexpected affinity interaction. The combination of post-Hartree-Fock calculation and density-functional theory (DFT) calculation reveals that oxygen vacancy defects on TiO<sub>2</sub> surface is more effective for CO<sub>2</sub> activation than stoichiometric TiO<sub>2</sub> surface, due to stronger interaction between CO<sub>2</sub> and TiO<sub>2</sub>.<sup>[31]</sup> A missing oxygen atom may leave two extra electrons at the site of the vacancy which reduce the adjacent two surface Ti<sup>4+</sup> sites to Ti<sup>3+</sup>, creating surface electron centers for the formation of the negatively charged CO<sub>2</sub><sup>•−</sup> species, in which the overlap between the (C-O)  $\pi^*$  anti-bonding orbitals of CO<sub>2</sub> and Ti 3d orbitals is favorable for electron transfer into CO<sub>2</sub>.

Oxygen-vacancy-rich ultrathin W<sub>18</sub>O<sub>49</sub> nanowires showed an excellent capability of selective reduction of CO<sub>2</sub> into CH<sub>4</sub> under irradiation with visible light irradiation ( $\lambda > 420$  nm).<sup>[184]</sup> The large number of oxygen vacancies existing in the nanowires, with a “trap” effect, can tightly “catch” CO<sub>2</sub> molecules and greatly enhance the adsorption of CO<sub>2</sub>. Oxygen vacancies being fully consumed after irradiation for 90 h could be easily recovered by utilizing the strong reducing power of NaBH<sub>4</sub>. Monoclinic phase Bi<sub>6</sub>Mo<sub>2</sub>O<sub>15</sub> microwires with post heating treatment at different temperatures from 600 °C to 800 °C showed increasing photocatalytic activity for reduction of CO<sub>2</sub> into CH<sub>4</sub>, resulting from the increasing concentration of surface oxide vacancies with the increase of post heating treatment.<sup>[181]</sup> The surface oxide vacancies can greatly prolong lifetime of photoexcited carriers which be beneficial for the separation of electron-hole pairs. However, higher temperature treatment of Bi<sub>6</sub>Mo<sub>2</sub>O<sub>15</sub> at 900 °C led to the damage of photocatalytic activity, owing to the disorder of atom arrangement and the loss of the crystallinity. Similar phenomenon was also observed over self-doped SrTiO<sub>3- $\delta$</sub>  with treatments in Ar at temperatures ranging from 1200 to 1400 °C for CO<sub>2</sub> reduction under visible light irradiation ( $\lambda > 420$  nm), in which the oxygen vacancies and Ti<sup>3+</sup> together induce an in-gap band to enhance the visible light absorption.<sup>[151]</sup> The sample treated at optimized temperature of 1300 °C realized the highest photocatalytic activity for CO<sub>2</sub> reduction with a quantum efficiency of 0.21% at 600 nm.

### 3.4. Macro-/Meso-/Microporous Structuralization

Porous materials usually have high surface areas, abundant surface states and readily accessed channels. Micro-/mesoporous photocatalysts for photocatalytic reduction of CO<sub>2</sub> are mainly classified two groups: i) photoactive species-dispersed zeolites,<sup>[42,44,48,51,52,56,57,63,70]</sup> mesoporous silica,<sup>[60,61,64,84,86,97,109,146]</sup> and metal-organic frameworks (MOFs),<sup>[197,246–248]</sup> and ii) pure mesoporous semiconductors, including TiO<sub>2</sub>,<sup>[96,100,112,115,137]</sup>

In(OH)<sub>3</sub>,<sup>[187]</sup> Bi<sub>2</sub>WO<sub>6</sub>,<sup>[186]</sup> Ga<sub>2</sub>O<sub>3</sub>,<sup>[189]</sup> ZnGa<sub>2</sub>O<sub>4</sub>,<sup>[191,202,249]</sup> and Zn<sub>2</sub>GeO<sub>4</sub>.<sup>[196,201]</sup>

Zeolites are representative micro/mesoporous materials that have been widely modified with TiO<sub>2</sub> photocatalysts for CO<sub>2</sub> reduction by taking advantage of their high surface area, high pore volume, tunable pore size, and good thermal and mechanical stability.<sup>[250]</sup> The role of the siliceous zeolites can be described as a so-called “catch and release effect of CO<sub>2</sub>”, i.e., i) the condensation of the reactants within the hydrophobic cavities of zeolites and ii) the efficient diffusion of the reactant onto the TiO<sub>2</sub> active sites.<sup>[251]</sup> Ti-containing zeolites shows improved CO<sub>2</sub> adsorption, reactivity, and selectivity of the products, including TS-1,<sup>[42,51]</sup> Ti-oxide/Y-zeolite,<sup>[44,51]</sup> Ti-MCM-41,<sup>[48,51]</sup> Ti-MCM-48,<sup>[48,51]</sup> Ti-FSM-16 mesoporous zeolite,<sup>[52]</sup> and Ti- $\beta$  zeolite.<sup>[56,57,63]</sup> The ex-Ti-oxide/Y-zeolite prepared by an ion-exchange method achieved a high photocatalytic efficiency and selectivity for the formation of CH<sub>3</sub>OH in the photocatalytic reduction of CO<sub>2</sub> with H<sub>2</sub>O, owing to that the titanium oxide species are highly dispersed within the zeolite cavities and exist in a tetrahedral coordination.<sup>[44]</sup> In contrast, CH<sub>4</sub> was major product using TiO<sub>2</sub> and imp-Ti-oxide/Y-zeolite as photocatalysts due to the involvement of aggregated octahedrally coordinated titanium oxide species. Compared with both Ti-MCM-41 with a large pore size (>20 Å) and 1D channel structure and TS-1 with smaller pore size (ca. 5.7 Å) and a three-dimensional (3D) channel structure, Ti-MCM-48 with a large pore size (>20 Å) and 3D channels showed higher photocatalytic efficiency and higher selectivity for the formation of CH<sub>3</sub>OH, which is due to a combined contribution of the high dispersion state of the Ti-oxide species and large pore size with a 3D channel structure.<sup>[51]</sup>

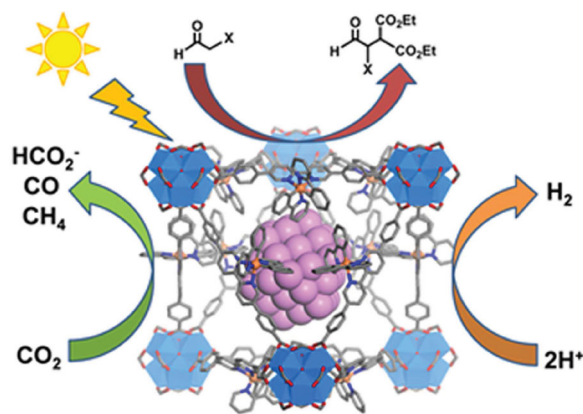
The hydrophilic-hydrophobic properties of these zeolites also prove to be a controlling factor in the selectivity for photocatalytic reduction of CO<sub>2</sub> with H<sub>2</sub>O.<sup>[56]</sup> Different concentrations of surface OH groups show different adsorption properties for H<sub>2</sub>O molecules toward the catalyst surface. Two types of Ti- $\beta$  zeolites by introducing OH<sup>−</sup> and F<sup>−</sup> ion during the prepared process possess hydrophilic and hydrophobic properties, respectively.<sup>[56]</sup> The selectivity for the formation of CH<sub>3</sub>OH over Ti- $\beta$ (F) with hydrophobic property was higher than that over Ti- $\beta$ (OH) with hydrophilic property. A typical self-standing Ti-containing porous silica thin film with a hexagonal pore structure exhibited higher photocatalytic activity for reduction of CO<sub>2</sub> into CH<sub>4</sub> and CH<sub>3</sub>OH with a high quantum yield of 0.28% than Ti-MCM-41 catalyst with the same pore structure in presence of water vapor under UV light irradiation.<sup>[60]</sup> Similar thin film with a cube pore structure and low concentration of surface OH groups showed higher selectivity for the formation of CH<sub>3</sub>OH.<sup>[61]</sup>

SBA-15, one well-known mesoporous silica with 1D hexagonal pore, has been studied as a catalyst support for photocatalytic reduction of CO<sub>2</sub>.<sup>[70,84,97,146]</sup> Ti-SBA-15 with an average pore diameter of 69 Å showed a 20-times increase of photocatalytic activity for reduction of CO<sub>2</sub> into CH<sub>4</sub> and CH<sub>3</sub>OH in the presence of H<sub>2</sub>O under UV light irradiation, compared with Ti-MCM-48 and Ti-MCM-41 with an average pore diameter of 27–29 Å, which was ascribed to higher dispersed titanium oxide species and larger pore size.<sup>[70]</sup> Ce-doped TiO<sub>2</sub> nanoparticles supported on SBA-15 mesoporous silica also remarkably enhanced photocatalytic activity for reduction of CO<sub>2</sub> into CO



and  $\text{CH}_4$  under UV–vis illumination, relative to  $\text{Ce-TiO}_2$ , due to the more effective diffusion and adsorption of  $\text{CO}_2$  through the ordered 1D pores in SBA-15.<sup>[146]</sup>

However, most of the reported zeolite and mesopore-based photocatalysts already investigated are only active in the UV region and their efficiency for  $\text{CO}_2$  reduction is still quite low. It is of great significance to develop highly efficient visible-light active photocatalysts. Metal–organic frameworks (MOFs) constructed by metal ions or small discrete clusters through the linkage of multidentate organic molecules are a new type of porous materials with crystalline frameworks.<sup>[252]</sup> The ability to organize different photoactive components in an MOF platform can make them comparable or even superior to the inorganic semiconductors to a certain degree to achieve artificial photosynthesis. The first MOF-based heterogeneous catalytic system was developed by doping different molecular complexes into UiO-67(Zr) for water oxidation, photocatalytic  $\text{CO}_2$  reduction, and organic transformations under visible light irradiation, which highlights the potential of using functionalized MOFs as visible light-responsive photocatalysts for  $\text{CO}_2$  reduction (Figure 7).<sup>[246]</sup> The introduction of  $\text{NH}_2$  groups into Ti-based MOF  $\text{Ti}_8\text{O}_8(\text{OH})_4(\text{bdc})_6$  (abbreviation as MIL-125(Ti)) to form  $\text{Ti}_8\text{O}_8(\text{OH})_4(\text{bdc-NH}_2)_6$  ( $\text{NH}_2$ -MIL-125(Ti)) allows extra absorption band in the visible light region with the absorption edge extending from 350 nm to around 550 nm, and an increase of adsorption capability toward  $\text{CO}_2$ .<sup>[247]</sup> The mechanism involves an electron transfer from the photoexcited organic ligand (visible-light-active 2-aminoterephthalate to the metal–oxo cluster  $[\text{TiO}_5(\text{OH})]$  in  $\text{NH}_2$ -MIL-125(Ti) to generate  $\text{Ti}^{\text{III}}$ , which is beneficial for the  $\text{CO}_2$  reduction under visible light irradiation.  $\text{NH}_2$ -UiO-66(Zr) exhibited higher activity for  $\text{CO}_2$  reduction than  $\text{NH}_2$ -MIL-125(Ti) in the presence of triethanolamine as sacrificial agent.<sup>[248]</sup> Zeolitic imidazolate frameworks (ZIFs), as a type of MOFs, have excellent thermal and chemical stability, and structural stability in water in addition to higher  $\text{CO}_2$  adsorption properties than zeolites.<sup>[253]</sup>  $\text{Zn}_2\text{GeO}_4/\text{ZIF-8}$  hybrid nanorods containing 25 wt% ZIF-8 exhibited 3.8-times higher dissolved  $\text{CO}_2$  adsorption capacity than the bare  $\text{Zn}_2\text{GeO}_4$  nanorods, resulting in a 62% enhancement in photocatalytic reduction of  $\text{CO}_2$  into liquid  $\text{CH}_3\text{OH}$  fuel.<sup>[197]</sup>



**Figure 7.** Schematic illustration of MOF-based heterogeneous catalytic system for water oxidation, photocatalytic  $\text{CO}_2$  reduction, and organic transformations. Reproduced with permission.<sup>[246b]</sup> Copyright 2011, American Chemical Society.

In addition to the large surface area, pure mesoporous semiconductors have high degree of crystallinity that leads to effective separation and transport of electron–hole pairs, and subsequently favorable for improved photocatalytic performance for  $\text{CO}_2$  reduction. A series of  $\text{TiO}_2$ -based mesoporous “french fries” (MFFs) ( $\text{TiO}_2/\text{ZnO}$ ,  $\text{TiO}_2/\text{Fe}_2\text{O}_3$ ,  $\text{TiO}_2/\text{CuO}$ ,  $\text{TiO}_2/\text{NiO}$ ,  $\text{TiO}_2/\text{Cr}_2\text{O}_3$ , and  $\text{TiO}_2/\text{CeO}_2$ ) were obtained by a facile furfural alcohol-derived polymerization–oxidation route. The production rate of  $\text{CO}_2$  to  $\text{CH}_4$  generated in the hybrid  $\text{TiO}_2/\text{ZnO}$  MFFs is at least 56 times higher than those obtained in the solid state reaction (SSR)- $\text{TiO}_2/\text{ZnO}$  and commercial P25, respectively. The high conversion efficiency results from: i) a faster diffusion transport of photogenerated electrons in hybrid  $\text{TiO}_2/\text{ZnO}$  than in pure  $\text{TiO}_2$ ; ii) the reduced recombination rate of photogenerated electron–hole pairs; iii) a higher specific surface area, which provides more adsorption of  $\text{CO}_2$  and  $\text{H}_2\text{O}$  vapor and more reaction active sites in the MFFs.<sup>[96]</sup> Mesoporous  $\beta\text{-Ga}_2\text{O}_3$  with an average pore size of 3.8 nm realized enhanced photocatalytic activity for reduction of  $\text{CO}_2$  into CO and  $\text{CH}_4$  than commercial  $\beta\text{-Ga}_2\text{O}_3$ , due to higher surface area and mesoporous channels for efficient  $\text{CO}_2$  adsorption.<sup>[189]</sup>

Mesoporous  $\text{ZnGa}_2\text{O}_4$  with a wormhole framework by an ion-exchange reaction at room temperature with 1 wt%  $\text{RuO}_2$  co-catalyst showed higher photocatalytic activity for reduction of  $\text{CO}_2$  into  $\text{CH}_4$  than  $\text{ZnGa}_2\text{O}_4$  obtained by SSR- $\text{ZnGa}_2\text{O}_4$ , owing to strong gas adsorption by the mesostructure and more reaction sites arising from high specific surface area ( $S_{\text{BET}} = 110.4 \text{ m}^2 \text{ g}^{-1}$  for meso- $\text{ZnGa}_2\text{O}_4$  and  $4.6 \text{ m}^2 \text{ g}^{-1}$  for SSR- $\text{ZnGa}_2\text{O}_4$ ).<sup>[191]</sup> This ion-exchange method can be extended to prepare  $\text{ZnGa}_2\text{O}_4$  nanocube<sup>[192]</sup> and other porous materials, such as  $\text{CoGa}_2\text{O}_4$ ,<sup>[191]</sup>  $\text{NiGa}_2\text{O}_4$ ,<sup>[191]</sup>  $\text{Zn}_2\text{GeO}_4$ ,<sup>[196,201]</sup> zinc gallogermanate solid solution,<sup>[237]</sup> and  $\text{Bi}_2\text{WO}_6$ .<sup>[186]</sup> Thus-prepared mesoporous  $\text{Zn}_2\text{GeO}_4$  showed higher photocatalytic activity for reduction of  $\text{CO}_2$  into  $\text{CH}_4$  ( $9.5 \text{ ppm g}^{-1} \text{ h}^{-1}$ ) compared with SSR- $\text{Zn}_2\text{GeO}_4$  ( $1.4 \text{ ppm g}^{-1} \text{ h}^{-1}$ ).<sup>[196]</sup> The conversion efficiency could be further improved to  $28.9 \text{ ppm g}^{-1} \text{ h}^{-1}$  with an apparent quantum yield of 0.2% at a wavelength of  $251 \pm 16 \text{ nm}$  when loading with 1 wt% Pt co-catalyst. Mesoporous  $(\text{Zn}_{1-x}\text{Ge}) (\text{N}_2\text{O}_x)$  by directly nitriding above mesoporous  $\text{Zn}_2\text{GeO}_4$  at high temperature under an  $\text{NH}_3$  atmosphere showed photocatalytic reduction of  $\text{CO}_2$  into  $\text{CH}_4$  under visible light irradiation ( $\lambda > 400 \text{ nm}$ ).<sup>[201]</sup>  $\text{ZnAl}_2\text{O}_4$ -modified mesoporous  $\text{ZnGaNO}$  solid solution synthesized by the combination of the ion-exchange method and nitriding treatment showed higher photocatalytic activity for  $\text{CO}_2$  reduction than bulk  $\text{ZnGaNO}$  solid solution under visible light irradiation ( $\lambda > 420 \text{ nm}$ ), attributed to the following three reasons: i) Constructing mesoporous structure improves the gas adsorption; ii)  $\text{ZnAl}_2\text{O}_4$  modification induces  $\text{CO}_2$  chemisorption on the photocatalyst; iii) the high Zn content in the  $\text{ZnAl}_2\text{O}_4$ -modified  $\text{ZnGaNO}$  narrows the band gap and extends the light absorption.<sup>[202]</sup>

While micro/mesoporous photocatalysts significantly enhance gas-phase reaction efficiency, however, gas diffusion limitation has still been a concerning problem. Ye et al. utilized 3D unique hierarchical structures of a green leaf as architecture-directing agents to fabricate “meso/macro porous perovskite titanates ( $\text{ATiO}_3$ , A = Sr, Ca, Pb) 3D” towards  $\text{CO}_2$  photo-reduction.<sup>[153]</sup> This methodology is desirable for development of a promising and ideal artificial photosynthesis system having

the analogous reaction features of natural photosynthesis by constructing: i) distinct 3D architecture for enhanced light harvesting; ii) efficient mass flow network of high surface areas and 3D multiple-scaled hierarchical porosity for suffering minimum gas diffusion and gas absorption; iii) appropriate catalysts with sufficiently negative CB for supplying strong reduction potential.

### 3.5. Crystal Facet Engineering

The surface atomic structure tunable by crystal facet engineering can adjust electronic band structure, surface energy and surface active sites, the adsorption of reactant, and desorption of reaction production.<sup>[254,255]</sup> Different surface structures have different selectivity for CO<sub>2</sub> adsorption and activation. Anatase TiO<sub>2</sub> is more favorable for CH<sub>4</sub> production, while rutile TiO<sub>2</sub> exhibits higher efficiency for the formation of CH<sub>4</sub> and CH<sub>3</sub>OH.<sup>[40]</sup> Furthermore, the yield of CH<sub>3</sub>OH formation over rutile TiO<sub>2</sub> was much higher on (100) than on (110), while the formation of CH<sub>4</sub> was only observed on (100) and but not on (110).<sup>[37]</sup> The reason is that (100) with a higher Ti/O surface ratio exhibits a more reductive property in the excited state than (110). Such a reductive surface allows a more facile reduction of CO<sub>2</sub> molecules especially for the formation of CH<sub>4</sub>. Single crystalline anatase TiO<sub>2</sub> rods with dominant reactive {010} facets showed a superior photocatalytic conversion of CO<sub>2</sub> into CH<sub>4</sub> relative to benchmark P25 TiO<sub>2</sub> nanocrystals.<sup>[91]</sup> The interaction of CO<sub>2</sub> on the (010) is predicted stronger than on both (101) and (001).<sup>[256]</sup> The ZnGa<sub>2</sub>O<sub>4</sub> nanocubes with exposed {100} facets exhibited improved performance in the photocatalytic reduction of CO<sub>2</sub> into CH<sub>4</sub> under UV-vis light irradiation, compared with mesoporous ZnGa<sub>2</sub>O<sub>4</sub> that has a larger specific surface area than ZnGa<sub>2</sub>O<sub>4</sub> nanocubes.<sup>[192]</sup> The light-hole effective mass on the {100} facets of ZnGa<sub>2</sub>O<sub>4</sub> nanocube corresponds to the high hole mobility, which contributes to the efficient H<sub>2</sub>O oxidation to offer the protons for promoting CO<sub>2</sub> reduction into CH<sub>4</sub>. The single crystal Zn<sub>2</sub>GeO<sub>4</sub> nanorods with dominant (110) crystal face also exhibited improved photocatalytic activity for CO<sub>2</sub> reduction, owing to the high specific surface area and dominant (110) crystal face.<sup>[195]</sup> The exposure of the {001} crystalline facet of the well-defined Bi<sub>2</sub>WO<sub>6</sub> nanoplate is a particular reactive surface energetically favoring the reduction of CO<sub>2</sub>.<sup>[185]</sup> Monoclinic BiVO<sub>4</sub> is more efficient than tetragonal BiVO<sub>4</sub> for the photocatalytic reduction of CO<sub>2</sub> into ethanol in aqueous solution, due to a more efficient migration of photogenerated electrons from the V 3d-block bands to the anchored CO<sub>2</sub> on the surface monoclinic phase than tetragonal phase.<sup>[178]</sup> For the monoclinic phase, a more strong asymmetry in the local environment around the Bi<sup>3+</sup> ion results in a stronger lone pair character in the Bi<sup>3+</sup> ion and hence a stronger tendency to form a Bi–O bond with CO<sub>2</sub>.<sup>[257]</sup>

It is interesting to note that sheet-like crystallites of molecular thickness and extremely high two-dimensional anisotropy can provide ratios of reactive sites on the surface that approach 100%. Recently, layered double hydroxides (LDHs) such as Zn–Al LDH,<sup>[258]</sup> Mg–In LDH,<sup>[259]</sup> and Zn–Cu–Ga LDH<sup>[258,260]</sup> are utilized for photocatalytic reduction of CO<sub>2</sub>. The versatility of the LDH structure allows tuning the compositions of metal

cations, charge balancing anions, and hydroxyl layer, which can adjust active sites, the adsorptive power of CO<sub>2</sub>, and tolerance of water. However, their perseverance of photocatalytic reduction of CO<sub>2</sub> needs to be examined due to the poor chemical stability of LDH. g-C<sub>3</sub>N<sub>4</sub> as a metal-free polymeric photocatalyst is considered as a promising visible-driven photocatalyst,<sup>[261]</sup> which has been applied for photocatalytic reduction for CO<sub>2</sub>.<sup>[262,263]</sup>

### 3.6. Utilization of Co-catalysts

Co-catalysts are known to play crucial role in the semiconductor-based photocatalysis.<sup>[240]</sup> Firstly, co-catalysts can promote the separation and migration of photogenerated charge carriers to reduce the recombination of electron–hole pairs, resulting from the formation of Schottky barrier between semiconductor and co-catalyst. In the case of metal nanoparticles supported on the semiconductor including Pt,<sup>[44,83,88,106,114]</sup> Rh,<sup>[38,54]</sup> Pd,<sup>[34,36,59,93,98]</sup> Cu,<sup>[39,62,69,72,73,80,83,84,86,94,95,102,110,116]</sup> Ag,<sup>[71,85,108,152,157]</sup> and Au,<sup>[137]</sup> the Fermi level of the metal normally lies below the CB edge of semiconductor, allowing the transfer of electrons from semiconductor to metal thermodynamically. Different valence states and particle size of co-catalysts influence the activity of photocatalytic reduction of CO<sub>2</sub>. An optimal 3wt% CuO loaded TiO<sub>2</sub> showed higher activity for CO<sub>2</sub> reduction than TiO<sub>2</sub> with 3wt% Cu<sub>2</sub>O or Cu, possibly due to lower potential redox of Cu<sup>2+</sup>/Cu<sup>+</sup> than that of Cu<sup>2+</sup>/Cu<sup>0</sup> or Cu<sup>+</sup>/Cu<sup>0</sup>.<sup>[116]</sup> Ultrafine Pt nanoparticles with optimized size of 0.5–2 nm promote TiO<sub>2</sub> films of 1D, single crystalline structure to display extremely high CO<sub>2</sub> photoreduction efficiency with highly selective formation of CH<sub>4</sub>. Either smaller or bigger one resulted in lower performance.<sup>[106]</sup> The reason is that smaller Pt nanoparticles possibly have higher energy band separation due to quantum confinement, restraining electron transfer from the CB of TiO<sub>2</sub> to Pt. As Pt nanoparticles become larger, electrons can transfer freely to Pt nanoparticles with an energy level lower than –4.4 eV (the lower energy level of the CB of TiO<sub>2</sub>). However, the properties of too large Pt nanoparticles may be approaching that of bulk Pt, capturing both photoelectrons and holes and acting as recombination centers. RuO<sub>2</sub> as oxidation site to collect holes is always loaded with other co-catalysts trapping electrons as reduction sites for photocatalytic reduction of CO<sub>2</sub>.<sup>[59,194,200]</sup> The photocatalytic activity of reduction of CO<sub>2</sub> to HCOO<sup>–</sup> improved after RuO<sub>2</sub> and Pd were co-deposited on the surface of TiO<sub>2</sub>, owing to that photogenerated electrons transfer from TiO<sub>2</sub> to Pd sites and photogenerated holes inject into RuO<sub>2</sub> sites.<sup>[59]</sup> Co-loading with 1 wt% Pt and 1 wt% RuO<sub>2</sub> on Zn<sub>2</sub>GeO<sub>4</sub><sup>[194]</sup> or Zn<sub>1.7</sub>GeN<sub>1.8</sub>O<sup>[200]</sup> apparently enhanced photocatalytic activity for reduction of CO<sub>2</sub> into CH<sub>4</sub>, even higher than the sum of the two individual co-catalysts, attributed to that Pt and RuO<sub>2</sub> can trap the electrons and holes respectively, hence inhibiting the recombination of electron–hole pairs.

Secondly, co-catalysts effectively lower the barrier for CO<sub>2</sub> activation, owing to their better conductivity and lower over potential. The activation energy of E<sub>a</sub> for commercial P25 and 3wt% CuO loaded TiO<sub>2</sub> is ca. +26 and +12 kJ/mol, respectively.<sup>[116]</sup> The apparent lower activation energy of the latter hence improved the efficiency of CO<sub>2</sub> reduction. CO<sub>2</sub> on Rh-loaded TiO<sub>2</sub> prefers to bond to the surface with both C (linked



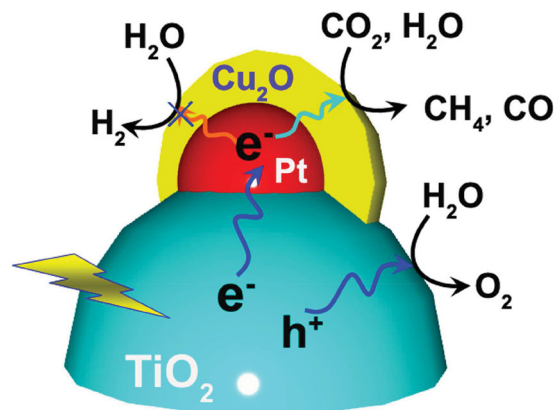
to a Rh atom) and one of the O atom (linked to the oxygen vacancy of  $\text{TiO}_2$ ), which leads to the cleavage of C-O bond easily under light irradiation.<sup>[26a]</sup> The addition of MgO onto Pt-loaded  $\text{TiO}_2$  significantly enhanced the activity of photocatalytic reduction of  $\text{CO}_2$  to  $\text{CH}_4$ , because the chemisorbed  $\text{CO}_2$  molecule on the MgO layer on  $\text{TiO}_2$  crystallites becomes destabilized and its reactivity is believed to be higher than that of the linear  $\text{CO}_2$  molecule.<sup>[26b]</sup>

Thirdly, co-catalysts can affect the selectivity of products in view of the formation of many different products during the process of  $\text{CO}_2$  reduction. Deposition of a series of 2 wt% metals (Pt, Rh, Pd, Cu, Ru, and Au) on  $\text{TiO}_2$  considerably accelerated photocatalytic reduction of  $\text{CO}_2$  to  $\text{CH}_4$  and/or acetic acid, and product distribution could be controlled by changing sorts of the metals deposited on  $\text{TiO}_2$ .<sup>[36]</sup> Deposition of  $\text{TiO}_2$  with Pt or Pd mainly produced  $\text{CH}_4$ , but a considerable amount of acetic acid was produced using other metals loaded  $\text{TiO}_2$ . NiO as an efficient co-catalyst was deposited on  $\text{InTaO}_4$  with a bandgap of 2.6 eV for selective reduction of  $\text{CO}_2$  into  $\text{CH}_3\text{OH}$  under visible light irradiation.<sup>[172,174,175]</sup>

Fourthly, the co-catalysts also play the role of inhibiting the back-recombination reaction,<sup>[241,264]</sup> including the re-oxidation (re-reduction) of hydrocarbon fuels (generated  $\text{O}_2$ ) and the splitting of  $\text{H}_2\text{O}$  into  $\text{H}_2$ . Perovskite  $\text{BaLa}_4\text{Ti}_4\text{O}_{15}$  plates with Ag co-catalyst on the edge exhibited high activity for photocatalytic reduction of  $\text{CO}_2$  into  $\text{CO}$ , compared with pure  $\text{BaLa}_4\text{Ti}_4\text{O}_{15}$  only for water splitting.<sup>[157]</sup> The reason is that the edge of a  $\text{BaLa}_4\text{Ti}_4\text{O}_{15}$  photocatalyst is the reduction site while the basal plane is the oxidation site for water splitting.  $\text{CO}_2$  reduction mainly proceeds on Ag co-catalyst loaded on the edge competing with  $\text{H}_2\text{O}$  reduction, while  $\text{O}_2$  evolution mainly carries on the basal plane. The separation of the reaction sites could also contribute to suppression of back reactions such as oxidation of formed  $\text{CO}$ .  $\text{Ni@NiO}$  and similar core-shells such as  $\text{Rh@Cr}_2\text{O}_3$  can prevent back reaction of the formation of  $\text{H}_2\text{O}$  from  $\text{H}_2$  and  $\text{O}_2$ .<sup>[16]</sup>  $\text{Rh}_{1.32}\text{Cr}_{0.66}\text{O}_3$  co-catalyst suppress the back reaction of reduction of  $\text{O}_2$  to  $\text{O}_2^-$ , compared with  $\text{Cu}_x\text{Ag}_y\text{In}_z\text{Zn}_k\text{S}$  solid solutions with  $\text{RuO}_2$  co-catalyst.<sup>[216]</sup>

The design of bifunctional co-catalysts with a proper structure is also crucial for obtaining high  $\text{CO}_2$  conversion activity. A core-shell-structured  $\text{Pt@Cu}_2\text{O}$  co-catalyst on  $\text{TiO}_2$  was designed to promote photocatalytic reduction of  $\text{CO}_2$  with  $\text{H}_2\text{O}$  to  $\text{CH}_4$  and  $\text{CO}$  (Figure 8). The Pt core extracts the photogenerated electrons from  $\text{TiO}_2$  and transfer the electrons to  $\text{Cu}_2\text{O}$  shell, and the  $\text{Cu}_2\text{O}$  shell provides sites for the preferential activation and conversion of  $\text{CO}_2$ .<sup>[114]</sup> Core-shell structure of metallic Ni core and NiO shell on the surface of  $\text{InTaO}_4$  by reduction-oxidation pretreatment also exhibited a positive effect on the catalyst for improving activity for  $\text{CO}_2$  reduction.<sup>[172]</sup> Ni core promotes the transfer of photogenerated electrons to NiO shell surface while NiO shell provides active sites to react with absorbed  $\text{CO}_2$ .<sup>[164,176]</sup>  $\text{Cu}_x\text{Pt}_y$  alloy co-catalysts coated on N-doped  $\text{TiO}_2$  nanotube arrays<sup>[83]</sup> or periodically modulated multiwalled  $\text{TiO}_2$  nanotube arrays<sup>[107]</sup> showed exhibited high catalytic activity for the photoreduction of  $\text{CO}_2$ , in compared with individual Cu and Pt. The optimized  $\text{Cu}_{0.33}\text{Pt}_{0.67}$  compositions were found to be the most reactive catalyst.<sup>[107]</sup>

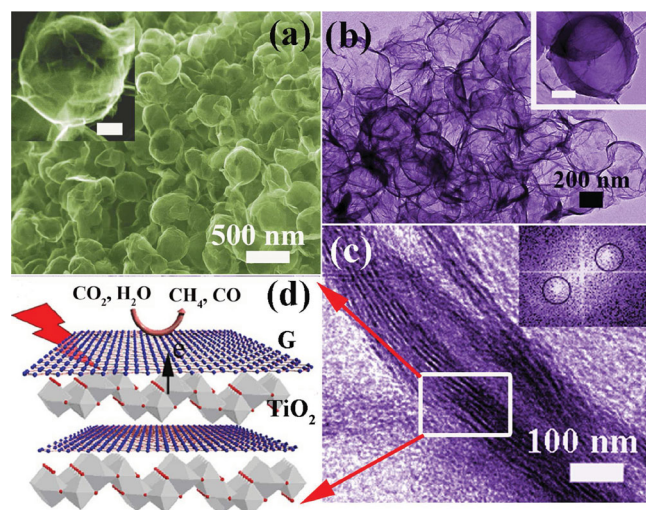
Fifthly, in addition to acting as electron sinker (or trapper), the surface plasmon resonance mediated-photocatalysis of



**Figure 8.** Schematic illustration of the  $\text{Pt@Cu}_2\text{O}$  core-shell co-catalyst on  $\text{TiO}_2$ .

noble metals (mainly Ag and Au) becomes a rising research star in the harvesting conversion of solar energy (especially in visible region) recently.<sup>[129,265]</sup> Au-deposited  $\text{TiO}_2$  film showed 24-times enhancement for  $\text{CO}_2$  reduction in the presence of water vapor compared with pure  $\text{TiO}_2$  under visible irradiation (532 nm), owing to the strong electric fields created by the surface plasmon resonance of the Au nanoparticles that excites electron-hole pairs locally in the  $\text{TiO}_2$  at a rate several orders of magnitude higher than the normal incident light.<sup>[129]</sup> Additional products including  $\text{CH}_4$ ,  $\text{C}_2\text{H}_6$ ,  $\text{CH}_3\text{OH}$ , and  $\text{HCHO}$  were obtained under UV irradiation (254 nm), because the transfer of plasmon-excited electrons from Au to  $\text{TiO}_2$  allows both the excited electrons in Au and  $\text{TiO}_2$  contribute to  $\text{CO}_2$  reduction with  $\text{H}_2\text{O}$  vapor.

Besides appropriate metals and metal oxides, carbon materials such as graphitic carbon,<sup>[115]</sup> carbon nanotube (CNT)<sup>[77,147]</sup> and graphene<sup>[90,104,111]</sup> can also act as electron acceptor and transporter to efficiently hinder recombination of electron-hole pairs, which play the role of co-catalyst for photocatalytic reduction of  $\text{CO}_2$ . Anatase  $\text{TiO}_2$  nanoparticles with diameter of 8–14 nm dispersed in the mesoporous graphitic carbon exhibited an excellent higher photocatalytic activity for reduction of  $\text{CO}_2$  into  $\text{CH}_4$  under simulated solar irradiation, compared with commercial P25.<sup>[115]</sup> CNT@Ni-doped  $\text{TiO}_2$  showed higher activity for reduction of  $\text{CO}_2$  into  $\text{CH}_4$  than Ni-doped  $\text{TiO}_2$  under visible light irradiation, mainly attributed to the transfer of electrons from  $\text{TiO}_2$  to CNTs through d-p electron orbital overlap between  $\text{TiO}_2$  and CNT.<sup>[147]</sup> Graphene offers an excellent electron-transport property and possesses an extremely high specific surface area for photocatalysis.<sup>[266]</sup> Robust hollow spheres consisting of  $\text{Ti}_{0.91}\text{O}_2$  nanosheets and graphene showed 9-times enhancement for photocatalytic  $\text{CO}_2$  reduction into dominant  $\text{CO}$  and minor  $\text{CH}_4$  than commercial P25 (Figure 9), owing to: i) the ultrathin nature of  $\text{Ti}_{0.91}\text{O}_2$  nanosheets could lead to rapid migration of charge carriers onto the surface for  $\text{CO}_2$  reduction; ii) the compact stacking of both two-dimensional nanosheets through surface conjugation (d- $\pi$  conjugate) enhances the lifetime of the charge carriers, due to the electron transfer from  $\text{Ti}_{0.91}\text{O}_2$  nanosheets to graphene; iii) the hollow structure potentially acts as a photon trap to enhance light absorption by multiscattering of incidence

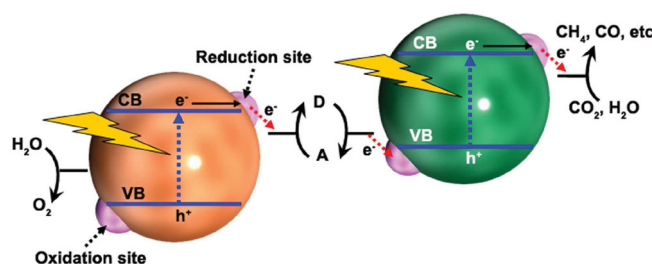


**Figure 9.** a) FE-SEM image and b,c) TEM image of  $(\text{G-Ti}_{0.91}\text{O}_2)_5$  hollow spheres. d) Schematic illustration of alternating  $\text{Ti}_{0.91}\text{O}_2$  nanosheets and G. Reproduced with permission.<sup>[104]</sup> Copyright 2012, Wiley-VCH.

light.<sup>[104]</sup> Graphene- $\text{TiO}_2$  hybrid nanosheets prepared by in situ simultaneous reduction-hydrolysis technique exhibited high and selective photocatalytic activity for  $\text{CO}_2$  reduction into  $\text{CH}_4$  and  $\text{C}_2\text{H}_6$ , which results from fast migration of electrons from  $\text{TiO}_2$  to graphene promoted by the formation of d- $\pi$  electron orbital overlap (Ti-O-C band).<sup>[111]</sup> The synergistic effect of the surface- $\text{Ti}^{3+}$  sites and graphene is beneficial for the generation of  $\text{C}_2\text{H}_6$ , and the yield of the  $\text{C}_2\text{H}_6$  increased with the content of incorporated graphene. Optimized graphene-P25 nanocomposites based on the less defective solvent-exfoliated graphene showed approximately 7-times improvement for reducing  $\text{CO}_2$  into  $\text{CH}_4$  than pure P25 under visible illumination.<sup>[90]</sup> The less-defective graphene, which possesses higher electrical mobility, facilitates the charge-carrier transport to surface sites to participate in the chemical reaction.

### 3.7. Z-Scheme Construction

The artificial Z-Scheme system, which imitates the natural photosynthesis of green plants for converting  $\text{CO}_2$  into carbohydrates, was designed for photocatalytic reduction of  $\text{CO}_2$  into hydrocarbon fuels.<sup>[222,223,267]</sup> The artificial Z-Scheme system consists of two different semiconductor photocatalysts with reversible redox mediators driven by a two-step photoexcitation, which has the merit of keeping electrons/holes with stronger reduction/oxidation abilities on different active sites (Figure 10). Arai et al. constructed a Z-scheme system for selective  $\text{CO}_2$  conversion to  $\text{HCOO}^-$  (formate) using a two-electrode configuration operated with no external electrical bias, which achieved a high selectivity for  $\text{HCOO}^-$  production (>70%) and a solar energy conversion efficiency of 0.03–0.04% under simulated sunlight irradiation (AM1.5).<sup>[222]</sup> In the two-electrode configuration, p-type  $\text{InP}/\text{Ru}$  complex polymer hybrid photocatalyst for  $\text{CO}_2$  reduction and  $\text{TiO}_2/\text{Pt}$  photocatalyst for  $\text{H}_2\text{O}$  oxidation were used as working electrode (photocathode) and counter electrode (photoanode), respectively. The enhancement



**Figure 10.** Schematic energy diagram of Z-scheme photocatalytic  $\text{CO}_2$  reduction for a two-step photoexcitation system. D and A indicate electron donating and accepting species, respectively.

of conversion efficiency and reaction selectivity was mainly ascribed to the electron-transfer process from the photocathode to the photoanode. Furthermore, a higher solar energy conversion efficiency was achieved by substituting  $\text{SrTiO}_3$  for  $\text{TiO}_2$  as photoanode, ascribed to the more negative potential of the CB minimum of  $\text{SrTiO}_3$  than  $\text{TiO}_2$  to facilitate electron transfer from CB of the photoanode to VB of the photocathode.<sup>[223]</sup> Ishitani et al. successfully synthesized the first visible-light-driven Z-scheme for heterogeneous photocatalytic reduction of  $\text{CO}_2$  under visible light irradiation ( $\lambda > 400 \text{ nm}$ ) by using a  $\text{Ru}(\text{II})$  dinuclear complex ( $\text{RuBLRu}'$ ) for  $\text{CO}_2$  reduction and Ag-loaded  $\text{TaON}$  for methanol oxidation.<sup>[267]</sup> Isotope experiments clearly showed that this hybrid photocatalyst mainly produced  $\text{HCOOH}$  from  $\text{CO}_2$  and  $\text{HCHO}$  from methanol, which could convert solar energy into chemical energy with a positive change in the Gibbs energy of  $83.0 \text{ kJ mol}^{-1}$ .

## 4. Challenges and Perspectives

The energy shortage and environmental related problems are becoming the worsening worldwide problems and the great challenges for human in the 21st century. The solar-driven reduction of  $\text{CO}_2$  to value-added chemical fuels is a promising avenue to impact the global carbon balance, and the potential rewards are enormous. The present goal is to synthesize photoactive materials able to chemically couple these light driven redox reactions together and achieve conversion efficiency and selectivity that exceeds Nature's photosynthesis. However, photoconversion of  $\text{CO}_2$  involves thermodynamically uphill, multielectron, multihole and multiproton processes occurring on a multicomponent photocatalyst. Longstanding challenges are presented in the fields of catalysis, energy science, semiconductor physics and engineering, and green chemistry.

To develop effective  $\text{CO}_2$  fixation and conversion, several key considerations must be balanced, including: i) a deep understanding of processes that occur on the surface of photocatalysts during  $\text{CO}_2$  reduction, e.g., adsorption/desorption of  $\text{CO}_2$  and intermediate products, as well as the role of adsorbed water. In this regard, the research would integrate experimental and computational approaches to evaluate catalyst activity toward  $\text{CO}_2$  photoreduction, and would advance fundamental understanding in the multidisciplinary areas of material science, surface chemistry, photocatalysis, and nanotechnology. Scanning tunneling microscopy (STM) would directly determine  $\text{CO}_2$  adsorption sites and probe molecular orbitals

and charge transfer. Density functional theory (DFT) calculations of the local density of states, lowest-energy molecular structure, and vibrational modes would theoretically clarify potential reaction mechanisms. The synergy of STM and DFT would provide a logical framework for future study aimed at building an atomic-scale understanding of CO<sub>2</sub> photoreduction; ii) potential great improvement of efficiency in CO<sub>2</sub> photoreduction can result from significantly increasing the lifetime of the charge-separated state. The time scale of this electron-hole recombination is two to three orders of magnitude faster than other electron-transfer processes. Therefore, any process that inhibits electron-hole recombination would greatly increase the efficiency and rates of CO<sub>2</sub> photoreduction. For example, electrical conductivity and diffusion length of the photocatalyst should be as high as possible to minimize recombination of electron-hole pairs. Ultrathin nanostructures may also facilitate the charge-carrier transport to surface reaction sites to participate reductive chemistry. Spatial separation of photoexcited electrons and holes can also be reduced the electron-hole recombination, which was achieved by co-catalysts, Z-scheme or heterostructures of coupling two semiconductors with properly aligned band structures; iii) the kinetics of CO<sub>2</sub> photoreduction is also dependent upon light absorption by the photocatalyst. Many optical techniques are potentially useful for harvesting light for improvement of the efficiency including: structuring for multiple light scattering to increase the effective optical path length, and up-conversion to transform non-absorbed infrared light to absorbed visible light; iv) high surface area and porosity are required to maximize the adsorption, transport and desorption of reactants, intermediates and products; v) as defects such as oxygen vacancies control most of the chemistry at many metal oxide surfaces, oxygen vacancies is believed to act as a very important role on electron trapping and activating CO<sub>2</sub>. The oxygen vacancies can be created either by doping with other anions or cations or by thermal treatments of stoichiometric photocatalysts, and these defects can be detected with in situ electron paramagnetic resonance spectroscopy, ultraviolet photoemission spectroscopy, and metastable impact electron spectroscopy and so on; vi) a high-efficiency process is also necessary to be founded on photoactive materials made of earth abundant, nontoxic, light-stable, scalable and low cost materials.

In addition, the efficiency of photocatalytic reduction of CO<sub>2</sub> could be deactivated after irradiation for long time. The deactivating phenomenon in the photocatalytic reduction of CO<sub>2</sub> can be attributed to the following three reasons: i) the adsorption or accumulation of intermediate products on the semiconductor surface could occupy photocatalytic reaction centers and also hinder the adsorption of CO<sub>2</sub> or H<sub>2</sub>O, which could lead the deactivation of semiconductors under the continuous irradiation; ii) the desorption of hydrocarbon production affects the continuous adsorption of reactants that take part in the photocatalytic reaction; iii) the surface contamination of semiconductors shields the light absorption, resulting in the reduction of the generation of electron-hole pairs. Therefore, it is necessary to pay more attention on the study of semiconductor deactivation in future work.

Nobel Laureate Jean Marie Lehn said "If it exists it can be synthesized". To learn nature's photosynthesis, which generates

everything "under the sun", it is sincerely hoped that multiple collaboration for this rapidly evolving field can lead to a breakthrough in the overall efficiency for the industrialization and commercialization with cost competitive with fossil fuels.

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