

# Supplementary Material for

# A parity-breaking electronic nematic phase transition in the spin-orbit coupled metal $Cd_2Re_2O_7$

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#### **Materials and Methods**

# S1. Sample Growth and Characterization

# Sample Growth

Single crystals of Cd<sub>2</sub>Re<sub>2</sub>O<sub>7</sub> were grown by vapor transport (41). X-ray diffraction measurements were performed on pulverized single crystals using a PANalytical X'Pert Pro powder x-ray diffractometer with Cu Kα radiation. No impurity peaks were observed. An elemental analysis was performed using a Hitachi TM-3000 scanning electron microscope equipped with a Bruker QUANTAX 70 energy dispersive x-ray system. The analysis confirmed an equal amount of Cd and Re within the resolution of the instrument. Magnetic susceptibility measurements were performed using a Quantum Design Magnetic Property Measurement System at temperatures ranging from 2 to 350 K. The results indicated high quality crystals without the presence of ReO<sub>2</sub> inclusions.

# Stoichiometric Origin of SHG Heterogeneity

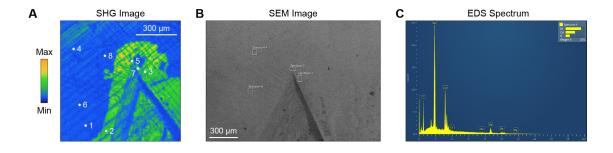
Wide-field SHG microscopy uncovered an inhomogeneous bulk response of the sample consisting of bright and dark regions. The boundaries between these regions are curved, tend to be near the edges of the sample, appear to be unaffected by structural domains, and have the same size scale as the single crystal itself. These facts strongly suggest chemical inhomogeneity during sample growth as a likely cause. To verify this hypothesis, we performed energy-dispersive x-ray spectroscopy (EDS) using an Oxford X-MaxN silicon drift detector attached to a Zeiss LEO 1550VP scanning electron microscope (SEM). This technique allows us to measure the spatially-resolved chemical composition of the sample near its surface. We collected EDS spectra at eight points—four inside a dark region and four inside a bright region—near a well-defined feature on the (111) sample surface, as shown in Fig. S1.

A typical EDS spectrum is shown in Fig. S1C. Peaks from O, Cd, and Re atoms can be identified, and fitting the spectrum allows us to infer the atomic percentages of the atoms at each sample location. Our results are summarized in the following table:

Location	Region	O atomic %	Cd atomic %	Re atomic %	Cd/Re ratio
1	Dark	61.85	19.17	18.98	1.0100
2	Bright	64.31	18.14	17.55	1.0336
3	Bright	62.39	19.08	18.53	1.0297
4	Dark	62.75	18.82	18.43	1.0212
5	Bright	62.52	19.08	18.40	1.0370
6	Dark	62.09	19.06	18.85	1.0111
7	Bright	60.86	19.73	19.41	1.0165
8	Dark	61.16	19.56	19.28	1.0145

Variability of oxygen stoichiometry is a known issue in the growth of  $Cd_2Re_2O_7$  (41) and is a prime candidate for the origin of the heterogeneity, but the oxygen atomic percentage does not appear to correlate with the bright and dark regions within the resolution of our experiment. We do detect, however, a difference in the average Cd/Re atomic ratio:  $1.029 \pm 0.009$  in the bright region and  $1.014 \pm 0.005$  in the dark region. A Student's *t*-test of the data shows that the difference is statistically significant (p = 0.027). The measured deviations of the Cd/Re ratio from 1 are within the *absolute* accuracy of the EDS instrument, so we cannot determine which of the two regions is more stoichiometric.

In oxide pyrochlores with chemical formula A<sub>2</sub>B<sub>2</sub>O<sub>7</sub>, the A and B cations can intermix under certain conditions and may even form a stable disordered phase (42). Indeed, in so-called "stuffed spin-ice" pyrochlores, the A/B atomic ratio can exceed 1.9 (43). Given our various observations as a whole, the most likely scenario is that inhomogeneous growth conditions cause Re atoms, with a much smaller ionic radius, to occupy some of the Cd sites in the darker regions of the sample. This will cause the nanoscale crystallinity to be slightly worse there (and may result in some localized metallic doping). It is then natural to expect the order parameter to be weaker in the more disordered regions of the sample, as we observe. For this reason, our study focuses on bright regions, where the bulk order parameters are larger with respect to the surface term.



**Fig. S1. Spatial mapping of sample stoichiometry.** (A) SHG image of the sample at 150 K near a well-defined feature on the surface. Eight sample points in bright and dark regions measured with EDS are indicated. (B) SEM image showing four of the rectangular regions used for EDS measurements. (C) Typical EDS spectrum, with x-ray peaks identified for O, Cd, and Re atoms.

# S2. Wide-Field SHG Microscopy

Wide-field SHG imaging was performed using ultrashort 100 fs optical pulses with a center wavelength of 800 nm produced at a 100 kHz repetition rate by a regeneratively-amplified Ti:sapphire laser system (Coherent Vitara-S and RegA 9050). The beam illuminated the entire (111) sample facet at an oblique 20° angle of incidence and with a fluence below 30  $\mu$ J cm<sup>-2</sup>. The reflected second-harmonic light was selected with a 400 nm narrow bandpass filter and linear polarizer, collected by an objective lens, and focused onto an EM-CCD camera (Andor iXon Ultra 897). Each pixel in the image captured a 6 × 6  $\mu$ m<sup>2</sup> area on the sample surface. Upon temperature cycling, we did not observe any movement of tetragonal domains, suggesting that they are pinned by static strain fields and other crystallographic defects.

#### S3. RA-SHG Measurements

RA-SHG measurements were performed with the same 800 nm light source as that used for the microscopy measurements. Both the sample and detector remained fixed. Rotation of the scattering plane was achieved by mechanically spinning transmissive optical elements about the central beam axis. A detailed description of the novel RA-SHG apparatus used can be found in Ref. 34. The laser was obliquely incident on the sample with a fixed 10° angle between the beam and the sample surface normal. The fluence of the beam was maintained at 600 µJ cm<sup>-2</sup>, with no noticeable degradation of the sample with time. The laser spot size on the sample was 30 µm full-width at half-maximum. Reflected second-harmonic light at 400 nm was selected with a narrow bandpass filter and measured with a two-dimensional EM-CCD (Andor iXon Ultra 897). Each complete RA pattern was acquired with a 30 s exposure time. Samples were measured in an optical cryostat with a vacuum pressure below 10<sup>-6</sup> Torr. We point out that an 800 nm wavelength ( $\hbar\omega = 1.5 \text{ eV}$ ) is resonant with interband transitions between occupied O 2p and unoccupied Re 5d states in Cd<sub>2</sub>Re<sub>2</sub>O<sub>7</sub> (44–46), significantly enhancing the SHG response of the crystal at this wavelength. We did not observe substantial steady-state heating of the sample based on the close agreement between  $T_c$  measured using SHG and  $T_c$  previously reported in the literature (26). This is consistent with a calculation of the heating amount ( $\sim 2$  K) using a thermal conductivity of 3 W m<sup>-1</sup> K<sup>-1</sup> (47).

# S4. Identification of All Six Domain Types

A cubic-to-tetragonal distortion will in general result in three types of domains, with each aligned along one of the three equivalent cubic lattice directions in the crystal. In addition, a parity-breaking distortion can occur in two separate ways related by inversion symmetry. Aided by SHG imaging, we are indeed able to find all six possible types of structural domains (three tetragonal × two parity) on the (111) sample surface. Performing RA-SHG measurements at each of the six domains yields distinct RA patterns, allowing us to fully distinguish them, as shown in Fig. S2.

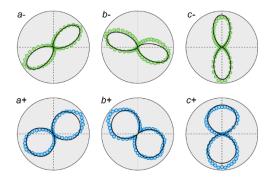


Fig. S2. Identification of domains via RA-SHG. Normalized  $S_{in}$ - $P_{out}$  RA-SHG patterns measured at 150 K at various locations on the sample surface showing all six possible structural domains. The three types of tetragonal domains (a, b, c) are distinguished by the directions of the RA lobes, which differ by 120°. The two types of parity domains (+, -) are distinguished by the shapes of the RA lobes, which differ because of interference between bulk and surface SHG.

## S5. RA-SHG Data for All Four Polarization Geometries

To minimize systematic errors, we use all four linear polarization geometries when fitting the nonlinear susceptibility to the data. At each temperature, all four RA patterns ( $P_{in}$ – $P_{out}$ ,  $P_{in}$ – $P_{out}$ ,  $P_{in}$ – $P_{out}$ , and  $P_{in}$ – $P_{out}$ , are measured and fit simultaneously with the *same* set of free parameters. Fig. S3 displays RA patterns and resulting fits for a representative selection of temperatures taken at (+) and (-) domains in dark regions of the sample, as well as a (-) domain in a bright region of the sample. Notably, there is no detectible bulk response for  $P_{in}$ – $P_{out}$ . We use this fact to constrain some of the bulk susceptibility components, as discussed in S6 below. Parity domains can be clearly distinguished from each other by the anisotropy resulting from interference with the surface SHG component. Because the bulk response is much stronger relative to the surface in the bright regions, our sensitivity to bulk symmetry breaking near  $P_c$  is increased there.

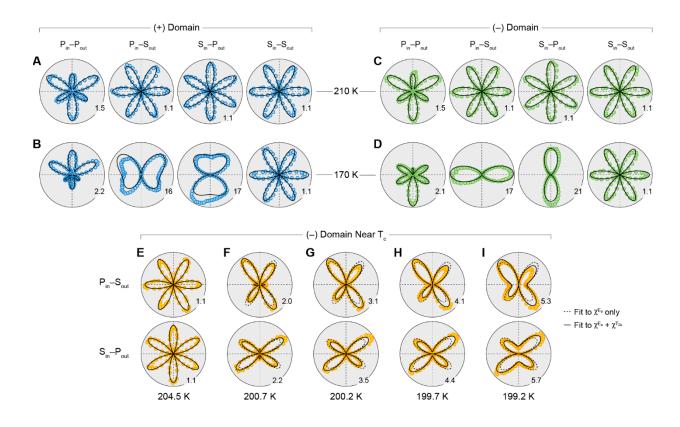


Fig. S3. RA-SHG domain contrast. (A)-(D) RA-SHG data with all four polarization geometries for a (+) domain (A, B) and a (-) domain (C, D) taken above  $T_c$  at 210 K (A and C) and below  $T_c$  at 170 K (B and D). At high temperatures, only the surface SHG contributes and the two domains are identical. Below  $T_c$ , interference between the static surface and growing bulk SHG contributions generates RA contrast between the parity domains. (E)-(I)  $P_{in}$ -S<sub>out</sub> and  $S_{in}$ -P<sub>out</sub> RA-SHG data taken at 204.5 K (E), 200.7 K (F), 200.2 K (G), 199.7 K (H), and 199.2 K (I). The loss of horizontal reflection symmetry in the RA curves is clearly detected below  $T_c$  but is not captured by fitting the data to a  $\overline{4}2m$  bulk point group (dashed curves). A  $\overline{4}$  bulk point group, however, fits the data well (solid curves).

# S6. Surface and Bulk Susceptibility Tensors

In our RA-SHG measurements, we observe a static, temperature-independent SHG component from inversion symmetry breaking at the sample surface. To isolate this component from the bulk response, we split the second-order optical susceptibility tensor that we measure into surface and bulk parts. Optical susceptibilities are complex. Our experiments, however, are only sensitive to relative phase differences between the susceptibility components, so we define the surface part to be strictly real but allow a complex phase for the bulk:  $\chi_{ijk}^{measured} = \chi_{ijk}^{S} + \chi_{ijk}^{B}$ .

In the cubic phase, the (111) surface of  $Cd_2Re_2O_7$  contains three mirror planes and a three-fold rotation axis and is represented by point group 3m. This symmetry group, together with SHG permutation symmetry, restricts the possible nonzero tensor elements of  $\chi_{ijk}^S$  to  $\chi_{xxz} = \chi_{xzx} = \chi_{yyz} = \chi_{yyz}$ ,  $\chi_{zxx} = \chi_{zyy}$ ,  $\chi_{yxx} = \chi_{xxy} = \chi_{xyx} = -\chi_{yyy}$ , and  $\chi_{zzz}$ , where coordinate axes are chosen so that one of the mirror planes is perpendicular to the x-axis and the surface normal is parallel to the z-axis (30). At high temperatures, where only  $\chi_{ijk}^S$  is nonzero, we find best agreement with our data when  $\chi_{xxz} = -0.9$ ,  $\chi_{zxx} = 0$ , and  $\chi_{zzz} = 200$ , in units where  $\chi_{yyy} = 1$ . We fix these values during our low-temperature fits and do not let them vary. In the y-axis labels of Fig. 4, " $|\chi^S|$ " is shorthand for the value of the  $\chi_{yyy}$  surface tensor element.

Our experiments show that  $\chi_{ijk}^B$  cannot be faithfully represented by a point group of higher symmetry than  $\bar{4}$ , whose nonzero tensor elements are  $\chi_{xxz} = \chi_{xzx} = -\chi_{yyz} = -\chi_{yzy}$ ,  $\chi_{xyz} = \chi_{xzy} = \chi_{yxz} = \chi_{yzx}$ ,  $\chi_{zxx} = -\chi_{zyy}$ , and  $\chi_{zxy} = \chi_{zyx}$  (30). Here, the coordinate axes x, y, and z are chosen to be aligned with the cubic crystallographic directions a, b, and c, respectively. As discussed in the main text, it is physically more appropriate to split this general bulk tensor into two parts, labeled  $\chi_{ijk}^{E_u}$  and  $\chi_{ijk}^{T_{2u}}$ , that transform like point groups  $\bar{4}2m$  (the isotropy group induced by  $E_u$  order) and  $\bar{4}m2$  (the isotropy group induced by  $T_{2u}$  order), respectively. Strictly speaking, these point groups are synonymous, differing by a trivial 45° rotation about the  $\bar{4}$  axis. Within a space group, however, the cubic crystallographic axes define fixed directions in space and the two become inequivalent. This decomposition of the  $\bar{4}$  tensor can be performed without loss of generality and unambiguously because  $\bar{4}2m$  tensors contain only  $\chi_{xyz} = \chi_{xzy} = \chi_{yzz} = \chi_{yzz}$  and  $\chi_{zxy} = \chi_{zyx}$  and  $\bar{4}m2$  tensors contain only  $\chi_{xxz} = \chi_{xzx} = -\chi_{yyz} = -\chi_{yzy}$  and  $\chi_{zxx} = -\chi_{zyy}$ . In particular, no tensor elements are shared between the two point groups.

We observe an additional symmetry in the  $\chi^B_{ijk}$  tensor:  $\chi_{zxy} = -2\chi_{xyz}$  and  $\chi_{zxx} = -2\chi_{xxz}$ . Experimentally, this symmetry is manifest in the complete absence of bulk SHG for the  $S_{in}$ - $S_{out}$  polarization geometry (see S5 above). For  $\chi^{E_u}_{ijk}$ , this constraint is directly related to the fact that a general  $\bar{4}2m$  tensor transforms like the direct sum  $E_u \oplus A_{2u}$ . By setting  $\chi_{zxy} = -2\chi_{xyz}$ , the  $A_{2u}$  part is projected out of the susceptibility tensor. For  $\chi^{T_{2u}}_{ijk}$ , the constraint  $\chi_{zxx} = -2\chi_{xxz}$  is not imposed by  $T_{2u}$  macroscopic symmetry considerations alone, but may follow from symmetries in the microscopic Hamiltonian. A further discussion of these additional symmetries can be found in S7 below.

In the end, only two independent complex parameters are allowed to vary with temperature in the susceptibility model that we employ:  $\chi_{xxz}$  and  $\chi_{xyz}$ . In the y-axis labels of Fig. 4, " $|\chi^{E_u}|$ " is shorthand for the magnitude of the  $\chi_{xyz}$  bulk tensor element and " $|\chi^{T_{2u}}|$ " is shorthand for the magnitude of the  $\chi_{xxz}$  bulk tensor element.

# S7. Linear Proportionality of Bulk Susceptibility and Odd-Parity Order Parameters

For a parity-breaking phase transition, we expect the bulk second-order optical susceptibility tensor to be linearly proportional to the order parameter for temperatures sufficiently close to  $T_c$ . The susceptibility, like the order parameter, is zero in the high-temperature centrosymmetric phase and nonzero in the low-temperature noncentrosymmetric phase. Furthermore, if the order parameter reverses sign, for example in different parity domains related by spatial inversion, so does the susceptibility. We have observed this sign reversal in our measurements and it allows us to distinguish (+) and (-) domains, as shown in S5 above. This means that a generic series expansion of the susceptibility in the order parameter will include only odd powers, with a linear term being the first allowed.

A more rigorous proof that the nonlinear optical susceptibility is linearly proportional to the oddparity order parameters may be obtained by expanding the Landau free energy up to lowest order in powers of components of the electric polarization **P** (a polar vector transforming like  $T_{1u}$ ) and the odd-parity order parameters  $\Psi_u$  (representing  $T_{2u}$  order) and  $\Phi$  (representing  $E_u$  order) (48):

$$F(\mathbf{P}) = F_0 + \frac{a}{2} P_i P_i - E_i P_i - b \Psi_u F_1(\mathbf{P}) - c \Phi F_2(\mathbf{P}) + \cdots,$$
  

$$F_1(\mathbf{P}) = (P_x^2 - P_y^2) P_z [1 - dP_z^2 / (P_x^2 + P_y^2 + P_z^2)],$$
  

$$F_2(\mathbf{P}) = P_x P_y P_z [1 - 3P_z^2 / (P_x^2 + P_y^2 + P_z^2)],$$

where summation over repeated indices is implied, **E** is the externally applied optical electric field, a > 0 because the system is not ferroelectric, b and c are constants coupling the electric polarization to the  $T_{2u}$  and  $E_u$  order parameters, respectively, and d is a dimensionless constant determined by the microscopic Hamiltonian of the system. Minimizing the free energy with respect to variation of the components of the electric polarization gives

$$P_i = \frac{1}{a}E_i + \left(\frac{b\Psi_u}{a}\right)\frac{\partial F_1}{\partial P_i} + \left(\frac{c\Phi}{a}\right)\frac{\partial F_2}{\partial P_i}.$$

By noting that  $P_i = \chi_{ij}E_j + \chi_{ijk}E_iE_k + \cdots$ , we may write

$$\chi_{ijk} = \frac{1}{2} \frac{\partial^2 P_i}{\partial E_j \partial E_k} \bigg|_{E=0} = \frac{1}{2a^2} \lim_{P_i \to 0} \lim_{P_j \to 0} \lim_{P_k \to 0} \frac{\partial^2 P_i}{\partial P_j \partial P_k}.$$

This formula gives  $\chi_{xyz} = \chi_{xzy} = \chi_{yxz} = \chi_{yzx} = -\chi_{zxy}/2 = -\chi_{zyx}/2 = c\Phi/2a^3$ , explicitly showing the linear proportionality of  $\chi^{E_u}$  and  $\Phi$  and confirming the additional symmetry constraint  $\chi_{zxy} = -2\chi_{xyz}$  discussed in S6 above. We also obtain  $\chi_{xxz} = \chi_{xzx} = -\chi_{yyz} = -\chi_{yzy} = b\Psi_u/a^3$  and  $\chi_{zxx} = -\chi_{zyy} = (1-d)b\Psi_u/a^3$ , showing the proportionality of  $\chi^{T_{2u}}$  and  $\Psi_u$ .

Interestingly, the empirical relation  $\chi_{zxx} = -2\chi_{xxz}$  that we observe in our measurements is achieved if d = 3, which is likely the result of a microscopic symmetry in the Hamiltonian of the system and offers a strong constraint on possible theories of the multipolar nematic phase in  $Cd_2Re_2O_7$ .

# **S8.** Landau Theory Symmetry Analysis

Within Landau's theory of second-order phase transitions, an order parameter  $\Phi$  that grows linearly with temperature in the low-symmetry phase,  $\Phi \propto (1-T/T_c)$  for  $T < T_c$ , must be secondary, induced by a coupling to a primary order parameter  $\Psi$  with mean field temperature dependence  $\Psi \propto \sqrt{1-T/T_c}$  (49). The so-called "faintness index"  $n \ge 2$  encodes the exponent of the primary order parameter that couples to the secondary order parameter in the Landau free energy, producing an invariant term of the form  $\Psi^n \Phi$  (50). Ferroic transitions that occur under these conditions are labeled "improper" (49–51). In particular, for a secondary order parameter with linear temperature dependence, n = 2.

A parity-breaking secondary order parameter with n=2 implies the existence of exactly two coupled primary order parameters, one that breaks inversion symmetry and one that preserves it. This is because a term like  $\Psi^2\Phi$  is not invariant under the parity operation if  $\Phi$  breaks inversion symmetry. In analogy with Mulliken notation, let us call the parity-even order parameter  $\Psi_g$  and the parity-odd order parameter  $\Psi_u$ . Then a coupling term like  $\Psi_g\Psi_u\Phi$  is invariant with respect to inversion symmetry and is allowed in the Landau free energy

$$F(T) = F_0 - \frac{a}{2} \left( 1 - \frac{T}{T_c} \right) \left( \Psi_g^2 + \Psi_u^2 \right) + \frac{b}{2} \Phi^2 - g \Psi_g \Psi_u \Phi + \frac{c}{4} \left( \Psi_g^4 + \Psi_u^4 \right) + \cdots,$$

where a, b, and c are positive constants to guarantee a stable solution and g controls the strength of the coupling between the primary and secondary order parameters. For convenience, we have taken the expansion coefficients of  $\Psi_g$  and  $\Psi_u$  to be identical ( $a_g = a_u$ ) and have ignored higher-order terms in the expansion; such simplifications are irrelevant to the arguments that follow. It should be noted, however, that the identical temperature dependence of the leading order coefficients for the primary order parameters—positive above  $T_c$  and negative below  $T_c$ —ensures that  $\Psi_g$  and  $\Psi_u$  become nonzero at the same temperature. Although not strictly guaranteed by theoretical arguments alone, such a condition is realized in practice if the coupled primary orders have the same physical origin. For example, the improper ferroelectric transitions observed in PbTiO<sub>3</sub>/SrTiO<sub>3</sub> superlattices and (Ca,Sr)<sub>3</sub>Ti<sub>2</sub>O<sub>7</sub> are driven by the simultaneous freezing of two phonon modes of different one-dimensional irreducible representations, one preserving inversion symmetry and the other breaking it (52, 53).

Minimization of the free energy with respect to variation of the order parameters gives

$$\Psi_g(T) = \Psi_u(T) = \sqrt{\frac{ab}{bc-g^2}} \sqrt{1 - \frac{T}{T_c}}, \quad \Phi(T) = \left(\frac{ag}{bc-g^2}\right) \left(1 - \frac{T}{T_c}\right).$$

By construction, we obtain a linear temperature dependence for the secondary order parameter  $\Phi$ . Higher-order terms in the free energy expansion or differing coefficients for  $\Psi_g$  and  $\Psi_u$  will, in general, change the constant factors in the above expressions but will not change the temperature dependence near  $T_c$ . This analysis shows that in Cd<sub>2</sub>Re<sub>2</sub>O<sub>7</sub>, the  $E_u$  structural distortion occurring at the 200 K phase transition (here represented by  $\Phi$ ) must be driven by a pair of primary order parameters  $\Psi_g$  and  $\Psi_u$ .

To proceed further, we impose four rules constraining the irreducible representations of the primary order parameters  $\Psi_q$  and  $\Psi_u$ .

- Rule #1: The Landau condition must hold, which states that for a second-order phase transition, no third-degree term can appear in the free energy expansion. This eliminates the  $A_{1g}$ ,  $E_g$ , and  $T_{2g}$  irreducible representations for  $\Psi_g$  (54).
- Rule #2:  $\Psi_u$  cannot transform like  $E_u$ . Otherwise, the term  $\Psi_u \Phi$  would occur in the free energy and the transition would not be improper ( $\Phi$  would have a square-root temperature dependence).
- Rule #3: In order for  $\Psi_g \Psi_u \Phi$  to be an allowed invariant in the free energy, the symmetric direct product of the irreducible representations for  $\Psi_g$  and  $\Psi_u$  must contain  $E_u$ .

These first three rules immediately imply that  $\Psi_g$  must transform like  $T_{1g}$ , and leave only two possibilities for the irreducible representation of  $\Psi_u$ :  $T_{1u}$  or  $T_{2u}$ . We now invoke the so-called "maximal isotropy group condition" (51), which states that because  $\Psi_g$  and  $\Psi_u$  transform like multidimensional irreducible representations, their directions in order parameter space must correspond to a maximal isotropy subgroup. Thus, we restrict our attention to the tetragonal isotropy subgroups that result from order parameter directions aligned with one of the cubic crystallographic axes. The  $E_u$  structural order parameter in  $Cd_2Re_2O_7$  is two-dimensional, but only the  $E_u^{(2)}$  partner is observed by structure-sensitive probes. This fact leads to a final rule that allows us to resolve the ambiguity in the symmetry of the  $\Psi_u$  order parameter.

• **Rule** #4: In the free energy expansion,  $\Psi_g \Psi_u$  cannot couple to the structural partner  $E_u^{(1)}$ . Otherwise, it would be finite below  $T_c$  and detected by experiment.

Only one pair of order parameter symmetries is consistent with this final rule:  $\Psi_g$  transforms according to  $T_{1g}$  in the  $\langle 001 \rangle$  direction [isotropy subgroup I4<sub>1</sub>/ $\alpha$  (48)] and  $\Psi_u$  transforms according to  $T_{2u}$  in the  $\langle 001 \rangle$  direction [isotropy subgroup I $\bar{4}2d$  (48)]. The maximal common subgroup of the order parameter isotropy groups is I $\bar{4}$ , which is the true space group of Cd<sub>2</sub>Re<sub>2</sub>O<sub>7</sub> in the low temperature phase.

We emphasize that this analysis, which is based solely on the observed linear temperature dependence of the  $E_u$  order parameter, necessitates the existence of a  $T_{2u}$  primary order parameter and therefore serves as a strong self-consistency check of our observations. This symmetry-based evidence of  $T_{2u}$  order also predicts the existence of a coupled  $T_{1g}$  primary order. The apparent coupling between the  $T_{2u}$  and  $T_{1g}$  primary orders suggests that they have a similar physical origin,

and we now discuss a possible mechanism for  $T_{1g}$  electronic order in  $\operatorname{Cd}_2\operatorname{Re}_2\operatorname{O}_7$  ( $T_{1g}$  transforms like a rotation, such as an angular momentum). Re<sup>5+</sup> ions have a  $5d^2$  valence configuration. Density functional theory calculations show that the octahedral crystal field strongly splits the 5d orbitals, pushing the unoccupied  $e_g$  states  $\sim 5$  eV higher in energy than the occupied  $t_{2g}$  states (44–46). If we consider a single  $\operatorname{Re}^{5+}$  ion in isolation, the spatial part of the two-valence-electron wave function will transform according to the direct product  $t_{2g} \otimes t_{2g} = A_{1g} + E_g + \left[T_{1g}\right] + T_{2g}$ , where the brackets denote the antisymmetric combination. According to Hund's first rule, electron interactions will favor the spin triplet state with term symbol  $^3T_{1g}$  (55). Although the preceding argument applies only to isolated ions, we anticipate that within  $\operatorname{Cd}_2\operatorname{Re}_2\operatorname{O}_7$  interactions among electrons in the  $\operatorname{Re} t_{2g}$  levels may favor a correlated triplet state with  $T_{1g}$  symmetry. A nuclear quadrupole resonance experiment indeed detected a moderate ferromagnetic enhancement, but no static magnetic order was observed (37), calling a simple spin triplet state into question. More theoretical work is necessary to identify a suitable physical mechanism for  $T_{1g}$  electronic order.

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