

# UNIVERSIDAD DE INVESTIGACION DE TECNOLOGIA EXPERIMENTAL YACHAY

Escuela de Ciencias Físicas y Nanotecnología

TÍTULO: First Principles and Machine Learning Investigations into the Atomic and Mechanical Properties of Cement Hydrates (C-S-H).

Trabajo de integración curricular presentado como requisito para la obtención del título de Físico

### **Autor:**

Balarezo J. Gabriel

### **Tutor:**

Ph.D - Pinto Henry

Urcuquí, Junio 2025

## **AUTORÍA**

Yo, Balarezo Balarezo Juan Gabriel, con cédula de identidad 0106019219, declaro que las ideas, juicios, valoraciones, interpretaciones, consultas bibliográficas, definiciones y conceptualizaciones expuestas en el presente trabajo; así cómo, los procedimientos y herramientas utilizadas en la investigación, son de absoluta responsabilidad de el/la autora (a) del trabajo de integración curricular. Así mismo, me acojo a los reglamentos internos de la Universidad de Investigación de Tecnología Experimental Yachay.

Urcuquí, junio 2025.

Juan Gabriel Balarezo Balarezo C.I: 0106019219

i

## AUTORIZACIÓN DE PUBLICACIÓN

Yo, Balarezo Balarezo Juan Gabriel, con cédula de identidad 0106019219, cedo a la Universidad de Investigación de Tecnología Experimental Yachay, los derechos de publicación de la presente obra, sin que deba haber un reconocimiento económico por este concepto. Declaro además que el texto del presente trabajo de titulación no podrá ser cedido a ninguna empresa editorial para su publicación u otros fines, sin contar previamente con la autorización escrita de la Universidad.

Asimismo, autorizo a la Universidad que realice la digitalización y publicación de este trabajo de integración curricular en el repositorio virtual, de conformidad a lo dispuesto en el Art. 144 de la Ley Orgánica de Educación Superior

Urcuquí, junio 2025.

Juan Gabriel Balarezo Balarezo C.I: 0106019219

For her, who is my everything.
For my family, who always supported me.

—Balarezo J. Gabriel

### Acknowledgments

Me gustaría expresar mi más sincero agradecimiento a mis padres, cuyo apoyo incondicional y aliento han sido la fuerza impulsora detrás de mi trayectoria académica, y cuyos sacrificios y confianza en mí han sido fundamentales para alcanzar este logro. A la Gnosis por enseñarme a mejorar como ser humano. A mis amigos, cuya compañía y amistad han traído alegría y risas en cada paso de este camino, gracias por siempre estar ahí. Un agradecimiento especial a mi novia, cuyo amor, comprensión y paciencia han sido una fuente constante de fortaleza e inspiración, siendo mi roca en los diversos desafíos. Me siento increíblemente afortunado de haber recibido la guía y el apoyo incondicional de mis co-asesores, María Camila y el Profesor Andrés, cuya experiencia, mentoría y dedicación han sido invaluables para la realización de esta tesis. Extiendo mi gratitud al mejor profesor de Yachay Tech, Henry Pinto.

Finalmente, dedico esta tesis a mi abuelo, cuya resiliencia y fortaleza frente a la adversidad han sido una fuente de inspiración. Su amor y sabiduría me han guiado a lo largo de los altibajos de la vida, y estoy eternamente agradecido por su presencia en mi vida.

Balarezo J. Gabriel

#### Resumen

Este estudio presenta una investigación exhaustiva de las propiedades magnéticas y electrónicas de los monocapas XGeTe<sub>3</sub> (X = Cr, Mn, Fe) y sus aleaciones aleatorias, utilizando la teoría del funcional de la densidad (DFT) con los funcionales PBE y PBESol, complementados con correcciones de Hubbard U. CrGeTe<sub>3</sub> exhibe un robusto orden ferromagnético (FM) con una brecha de banda calculada que concuerda bien con los datos experimentales, destacando su potencial para aplicaciones prácticas. La monocapa MnGeTe<sub>3</sub> muestra un comportamiento de medio metal (HM), lo que lo hace particularmente prometedor para aplicaciones espintrónicas. En contraste, FeGeTe<sub>3</sub> revela un estado fundamental antiferromagnético (AFM) y posibles inestabilidades dinámicas, lo que requiere una mayor exploración para optimizar sus propiedades electrónicas.

En las aleaciones aleatorias, se observaron modificaciones significativas en los momentos magnéticos y las estructuras electrónicas. Específicamente, en  $Cr_{1-x}GeMn_xTe_3$ , el desorden en el momento magnético sugiere estados magnéticos fundamentales complejos, beneficiosos para la espintrónica. Mientras tanto,  $Cr_{1-x}GeFe_xTe_3$  demuestra efectos de fuerte hibridación, indicando su idoneidad para sensores magnéticos y aplicaciones termoeléctricas. El sistema  $Fe_{1-x}GeMn_xTe_3$  revela desafíos estructurales a mayores concentraciones de Mn, pero su estabilidad termodinámica respalda su potencial uso en uniones túnel magnéticas.

En general, esta investigación resalta las considerables perspectivas tecnológicas de los monocapas XGeTe<sub>3</sub> y sus aleaciones aleatorias, abogando por un control preciso de las propiedades magnéticas y electrónicas para avanzar en sus aplicaciones en espintrónica, memorias magnéticas y dispositivos termoeléctricos.

Palabras clave: Teoría del Funcional de la Densidad, monocapas, ferromagnetismo, antiferromagnetismo, semi-metal, aleaciones aleatorias.

#### Abstract

This study presents a thorough investigation of the magnetic and electronic properties of XGeTe<sub>3</sub> (X = Cr, Mn, Fe) monolayers and their random alloys, utilizing density functional theory (DFT) with PBE and PBESol functionals, supplemented by Hubbard U corrections. CrGeTe<sub>3</sub> exhibits robust ferromagnetic (FM) ordering with a calculated band gap that aligns well with experimental data, highlighting its potential for practical applications. The MnGeTe<sub>3</sub> monolayer shows half-metallic (HM) behavior, making it particularly promising for spintronic applications. In contrast, FeGeTe<sub>3</sub> reveals an antiferromagnetic (AFM) ground state and potential dynamical instabilities, necessitating further exploration to optimize its electronic properties.

For the random alloys, significant modifications in magnetic moments and electronic structures were noted. Specifically, in  $Cr_{1-x}GeMn_xTe_3$ , magnetic moment disorder suggests complex magnetic ground states beneficial for spintronics. Meanwhile,  $Cr_{1-x}GeFe_xTe_3$  demonstrates strong hybridization effects, indicating its suitability for magnetic sensors and thermoelectric applications. The  $Fe_{1-x}GeMn_xTe_3$  system reveals structural challenges at higher Mn concentrations, but its thermodynamic stability supports its potential use in magnetic tunneling junctions.

Overall, this research emphasizes the considerable technological prospects of XGeTe<sub>3</sub> monolayers and their random alloys, advocating for precise control over magnetic and electronic properties to advance their applications in spintronics, magnetic memory, and thermoelectric devices.

**Keywords:** Density Functional Theory, monolayers, ferromagnetism, anti-ferromagnetism, half-metal, random alloys.

## Contents

Contents					
Li	st of	igures ix			
Li	st of	Tables	x		
1	Inti	roduction	1		
	1.1	Background	1		
	1.2	Problem Statement	2		
	1.3	General and Specific Objectives	2		
	1.4	Overview	2		
2	The	eoretical Background	4		
	2.1	Many Body Schrödinger Equation	4		
		2.1.1 The Coulomb Interaction	4		
		2.1.2 The Time-Independent Schrödinger Equation	5		
	2.2	The Born-Oppenheimer Approximation	7		
	2.3	Hartree-Fock Approximation	8		
		2.3.1 The Pauli Exclusion Principle	10		
		2.3.2 The Hartree-Fock Equations	10		
	2.4	Density Functional Theory	11		
		2.4.1 First Hohenberg-Kohn Theorem	12		
		2.4.2 Second Hohenberg-Kohn Theorem	13		
		2.4.3 Kohn-Sham Equations	14		
		2.4.4 Exchange-Correlation Functionals	15		
		2.4.4.1 Local Density Approximation	15		
		2.4.4.2 Generalised Gradient Approximation	16		
		2.4.4.3 Hybrid Functionals	17		
	2.5	Density Functional Tight Binding	19		
	2.6	Ab initio Molecular Dynamics	19		
		2.6.1 Born-Oppenheimer Molecular Dynamics	19		
		2.6.2 Car-Parrinello Molecular Dynamics	19		
	2.7	Computational Implementation in VASP	19		
		2.7.1 Pseudopotentials	19		
		2.7.2 K-Point Sampling and Cutoff Energy	19		
		2.7.3 Projector Augmented Wave Method	19		
		2.7.4 Machine Learning Interatomic Potentials	19		

Co	Contents		
3	Methodology	20	
4	Results & Discussion	21	
	4.1 My results	21	
	4.1 My results	2	
B	bliography	22	

## List of Figures

1.1	A four-level model representing the upscaling of C-S-H properties from the nanoscale to the engineering scale. (a) snapshot of C-S-H's nanostructure. (b) microstructure of C-S-H created by agglomeration of randomly oriented C-S-H nanoparticles. (c) microtexture of hardened paste composed by hydration products. (d) macrotexture of cement paste at the engineering scale. Adapted from <sup>9</sup>	2
2.1	Jacob's ladder of exchange-correlation functionals as proposed by J. Perdew <sup>28</sup> . The ladder categorises the functionals into rungs, from the simplest approximation (LDA) at the bottom progressing to more sophisticated and accurate approximations (Generalised Random Phase) at the uppermost rung	
4.1	A schematic representation of the DFT formalism. The many-body wavefunction is replaced by a single-particle wavefunction, which is used to calculate the electron density. The electron density is then used to calculate the total energy of the system. Adapted from 10	91

## List of Tables

## Chapter 1

## Introduction

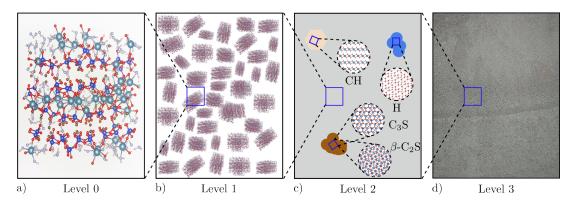
## 1.1 Background

Concrete is the synthetic material currently produced in volumes larger than any other material on Earth. With an annual consumption of approximately 35 billion tonnes, it is only second to water in terms of global usage<sup>1,2</sup>. As the backbone of modern infrastructure, it provides the foundations for buildings, bridges, roads, dams, and other structures essential for societal development. Its widespread adoption arises from a unique combination of strength, versatility, and cost-effectiveness<sup>3</sup>, rendering it indispensable to the construction industry.

Nevertheless, despite the ubiquity of concrete, the properties of its key constituent, cement, remain incompletely understood. Cement is a chemically complex material, composed of a heterogeneous mixture of minerals that undergo a series of hydration reactions upon contact with water. The principal product of cement hydration—and the primary binding phase of concrete—calcium silicate hydrate (C-S-H)<sup>1</sup> is the responsible for the mechanical strength, chemical and transport properties and durability of hardened cement paste and, consequently, of concrete itself<sup>4,5,6,7</sup>. Therefore, understanding the atomic and mechanical properties of C-S-H is of the uttermost importance to better formulate cementitious materials with enhanced performance and durability.<sup>8</sup>

In the last two decades, significant efforts have been made to

<sup>&</sup>lt;sup>1</sup>Cement chemistry nomenclature



**Figure 1.1:** A four-level model representing the upscaling of C-S-H properties from the nanoscale to the engineering scale. (a) snapshot of C-S-H's nanostructure. (b) microstructure of C-S-H created by agglomeration of randomly oriented C-S-H nanoparticles. (c) microtexture of hardened paste composed by hydration products. (d) macrotexture of cement paste at the engineering scale. Adapted from <sup>9</sup>.

## 1.2 Problem Statement

Nunc posuere quam at lectus tristique eu ultrices augue venenatis. Vestibulum ante ipsum primis in faucibus orci luctus et ultrices posuere cubilia Curae; Aliquam erat volutpat. Vivamus sodales tortor eget quam adipiscing in vulputate ante ullamcorper. Sed eros ante, lacinia et sollicitudin et, aliquam sit amet augue. In hac habitasse platea dictumst.

## 1.3 General and Specific Objectives

Morbi rutrum odio eget arcu adipiscing sodales. Aenean et purus a est pulvinar pellentesque. Cras in elit neque, quis varius elit. Phasellus fringilla, nibh eu tempus venenatis, dolor elit posuere quam, quis adipiscing urna leo nec orci. Sed nec nulla auctor odio aliquet consequat. Ut nec nulla in ante ullamcorper aliquam at sed dolor. Phasellus fermentum magna in augue gravida cursus. Cras sed pretium lorem. Pellentesque eget ornare odio. Proin accumsan, massa viverra cursus pharetra, ipsum nisi lobortis velit, a malesuada dolor lorem eu neque.

### 1.4 Overview

Sed ullamcorper quam eu nisl interdum at interdum enim egestas. Aliquam placerat justo sed lectus lobortis ut porta nisl porttitor. Vestibulum mi dolor, lacinia molestie gravida at, tempus vitae ligula. Donec eget quam sapien, in viverra eros. Donec pellentesque justo a massa fringilla non vestibulum metus vestibulum. Vestibulum in orci quis felis tempor lacinia. Vivamus ornare ultrices facilisis. Ut hendrerit volutpat vulputate. Morbi condimentum venenatis augue, id porta ipsum vulputate in. Curabitur luctus tempus justo. Vestibulum risus lectus, adipiscing

nec condimentum quis, condimentum nec nisl. Aliquam dictum sagittis velit sed iaculis. Morbi tristique augue sit amet nulla pulvinar id facilisis ligula mollis. Nam elit libero, tincidunt ut aliquam at, molestie in quam. Aenean rhoncus vehicula hendrerit.

## Chapter 2

## Theoretical Background

This chapter is devoted to present the theoretical foundations and formalism of Density Functional Theory (DFT) and related methods required for the development of the results presented in this work. Starting with the many-body Schrödinger equation, this chapter covers the Born-Oppenheimer approximation, the Hartree-Fock approximation, Hohenberg-Kohn theorems, the Kohn-Sham equations, exchange-correlation functionals, and definitions on Ab initio molecular dynamics (AIMD) and machine learning force fields (MLFFs), along with implementation details in the Vienna Ab initio Simulation Package (VASP).

Ultimately, this chapter aims to provide a comprehensive understanding of the theoretical framework that underpins the computational methods utilised in this work, enabling the reader to grasp the principles and assumptions that govern the simulations and analyses performed throughout.

## 2.1 Many Body Schrödinger Equation

In the realm of materials science, comprehending particle behaviour within a system inevitably demands resorting to intricate principles of quantum mechanics. Our journey shall commence by describing the physical laws that shape the interactions among particles constituting a system—electrons and nuclei alike.

#### 2.1.1 The Coulomb Interaction

Materials may be thought of as complex assemblies of electrons and nuclei, held together by a delicate balance between attractive Coulomb interactions—primarily between electrons and nuclei—and repulsive interactions between like-charged particles, such as electron-electron and

nucleus-nucleus pairs, which govern the overall dynamics of the material system<sup>10,11,12</sup>. From classical electrostatics, these interactions can be mathematically expressed as follows:

• Electron-electron interactions

$$\hat{V}_{ee} = \frac{1}{2} \sum_{i \neq j} \frac{e^2}{4\pi\epsilon_0 |\mathbf{r}_i - \mathbf{r}_j|}$$
(2.1)

• Electron-nucleus interactions

$$\hat{V}_{nn} = \frac{1}{2} \sum_{I \neq J} \frac{e^2}{4\pi\epsilon_0} \frac{Z_I Z_J}{|\mathbf{R}_I - \mathbf{R}_J|}$$
(2.2)

• Electron-nuclei interactions

$$\hat{V}_{en} = -\sum_{i \neq I} \frac{e^2}{4\pi\epsilon_0} \frac{Z_I}{|\mathbf{r}_i - \mathbf{R}_I|}$$
(2.3)

where e is the electronic charge,  $\epsilon_0$  is the vacuum permittivity,  $Z_I$  and  $Z_J$  are the atomic numbers of nuclei I and J, respectively, and  $\mathbf{r}_i$  and  $\mathbf{R}_I$  are the position vectors of electrons and nuclei, respectively. Moreover, we must also consider the kinetic energy of the collection of electrons and nuclei

$$\hat{T} = -\sum_{i} \frac{\hbar^2}{2m_e} \nabla_i^2 - \sum_{I} \frac{\hbar^2}{2M_I} \nabla_I^2$$
 (2.4)

where  $\hbar$  is the reduced Planck's constant,  $m_e$  is the electron mass, and  $M_I$  is the mass of the nucleus I.

### 2.1.2 The Time-Independent Schrödinger Equation

The Time-Independent Schrödinger Equation (TISE) lies at the heart of non-relativistic quantum mechanics, providing a mathematical framework to describe stationary electronic states of quantum systems. It takes the following form:

$$\hat{H}\psi(\mathbf{r}) = E\psi(\mathbf{r}) \tag{2.5}$$

where  $\hat{H}$  is the Hamiltonian of the system, encompassing both kinetic and potential energies,  $\psi(\mathbf{r})$  is the wavefunction of the system, and E is the energy eigenvalue associated with the state described by  $\psi(\mathbf{r})$ . It is important to note that Equation 2.5 is only applicable to a single particle, yet a material system is composed of many electrons (N) and nuclei (M) with spatial coordinates  $\mathbf{r}_1, \mathbf{r}_2, \ldots, \mathbf{r}_N$  and  $\mathbf{R}_1, \mathbf{R}_2, \ldots, \mathbf{R}_M$ , respectively. Therefore, we must introduce a

so-called many-body wavefunction given by:

$$\Psi(\mathbf{r}_1, \mathbf{r}_2, \dots, \mathbf{r}_N, \mathbf{R}_1, \mathbf{R}_2, \dots, \mathbf{R}_M) \tag{2.6}$$

On this basis, the many-body version of Equation 2.5 shall be constructed by combining the kinetic (Equation 2.4) and potential (Equations 2.1, 2.2, and 2.3) energy contributions, leading to the following expression:

$$\left[ -\sum_{i} \frac{\hbar^{2}}{2m_{e}} \nabla_{i}^{2} - \sum_{I} \frac{\hbar^{2}}{2M_{I}} \nabla_{I}^{2} + \frac{1}{2} \sum_{i \neq j} \frac{e^{2}}{4\pi\epsilon_{0} |\mathbf{r}_{i} - \mathbf{r}_{j}|} + \frac{1}{2} \sum_{I \neq J} \frac{e^{2}}{4\pi\epsilon_{0}} \frac{Z_{I}Z_{J}}{|\mathbf{R}_{I} - \mathbf{R}_{J}|} - \sum_{i,I} \frac{e^{2}}{4\pi\epsilon_{0}} \frac{Z_{I}}{|\mathbf{r}_{i} - \mathbf{R}_{I}|} \right] \Psi = E_{\text{tot}} \Psi$$
(2.7)

Equation 2.7 provides a complete description of the stationary states of a non-relativistic many-body system, under time-independent conditions and in the absence of external fields. Additionally, we can achieve a more compact formulation by introducing the concept of atomic units. To this end, let us consider the simplest electron-nucleus system—the hydrogen atom—where the electron orbital has an average radius  $a_0 \approx 0.529 \,\text{Å}$ . Thereby, the Coulomb energy for such a system is given by:

$$E_{\rm Ha} = \frac{e^2}{4\pi\epsilon_0 a_0} \tag{2.8}$$

where 'Ha' stands for Hartree. Within this framework, the Hartree energy represents the Coulomb interaction between two fundamental charges separated by a distance of one Bohr radius  $(a_0)$ . Moreover, the angular momentum quantisation condition for the electron in the hydrogen atom is given by

$$m_e v a_0 = \hbar \tag{2.9}$$

Additionally, the equilibrium condition between the nuclear attraction and the electron's centrifugal force can be expressed as:

$$\frac{e^2}{4\pi\epsilon_0 a_0^2} = \frac{m_e v^2}{a_0} \tag{2.10}$$

By combining Equations 2.8, 2.9, and 2.10, we derive the following relationships:

$$\frac{e^2}{4\pi\epsilon_0 a_0} = \frac{\hbar^2}{m_e a_0^2} \tag{2.11}$$

$$\frac{1}{2}m_e v^2 = \frac{1}{2}E_{\text{Ha}} \tag{2.12}$$

This latter relation showcases that the kinetic energy is of the same order as  $E_{\text{Ha}}$ , rendering it convenient to normalise Equation 2.7 by this quantity:

$$\left[ -\sum_{i} \frac{1}{2} a_{0}^{2} \nabla_{i}^{2} - \sum_{I} \frac{1}{2} \frac{1}{(M_{I}/m_{e})} \nabla_{I}^{2} + \frac{1}{2} \sum_{i \neq j} \frac{a_{0}}{|\mathbf{r}_{i} - \mathbf{r}_{j}|} + \frac{1}{2} \sum_{I \neq J} Z_{I} Z_{J} \frac{a_{0}}{|\mathbf{R}_{I} - \mathbf{R}_{J}|} - \sum_{i,I} Z_{I} \frac{a_{0}}{|\mathbf{r}_{i} - \mathbf{R}_{I}|} \right] \Psi = \frac{E_{\text{tot}}}{E_{\text{Ha}}} \Psi$$
(2.13)

A final simplification involves setting our energy units to Ha, distance units to  $a_0$ , and mass units to  $m_e$ . The last missing constant e is set to 1, leading to the following expression:

$$\left[ -\sum_{i} \frac{\nabla_{i}^{2}}{2} - \sum_{I} \frac{\nabla_{I}^{2}}{2M_{I}} + \frac{1}{2} \sum_{i \neq j} \frac{1}{|\mathbf{r}_{i} - \mathbf{r}_{j}|} + \frac{1}{2} \sum_{I \neq J} \frac{Z_{I}Z_{J}}{|\mathbf{R}_{I} - \mathbf{R}_{J}|} - \sum_{i,I} \frac{Z_{I}}{|\mathbf{r}_{i} - \mathbf{R}_{I}|} \right] \Psi = E_{\text{tot}} \Psi$$
(2.14)

Ultimately, even though Equation 2.14 provides an exact method capable of yielding various properties of a material system—such as elastic, thermal, and electronic properties—a combination of mathematical complexity and computational limitations renders it intractable to solve for any realistic system. Moreover, the wavefunction contains vastly more information than is necessary to describe most observable properties of a material. Therefore, we must resort to alternative formulations that allow us to extract only the relevant information from the wavefunction whilst reducing the computational cost of the calculations. The remainder of this chapter is dedicated to presenting such alternative approaches that ultimately lead to the computational methods employed throughout this work.

## 2.2 The Born-Oppenheimer Approximation

For atoms in a solid, we can think of nuclei as being held immobile in a fixed position, while electrons instantaneously react to any nucleus's movement. This assumption is based on the fact that nuclei are much heavier than electrons—by three to four orders of magnitude—making the former behave like classical particles. Thereby, we can rewrite the many-body wavefunction as a product of two wavefunctions:

$$\Psi(\mathbf{r}_1, \mathbf{r}_2, \dots, \mathbf{r}_N, \mathbf{R}_1, \mathbf{R}_2, \dots, \mathbf{R}_M) = \psi_{\mathbf{R}}(\mathbf{r}_1, \mathbf{r}_2, \dots, \mathbf{r}_N) \chi(\mathbf{R})$$
(2.15)

where  $\psi_{\mathbf{R}}$  is the electronic wavefunction parametrised by the nuclear positions  $\mathbf{R}$ , and  $\chi$  is the nuclear wavefunction. Furthermore, this significant mass disparity enables a systematic

approximation scheme, in which the electronic wavefunction is solved for fixed nuclei, and its solution is used as an effective potential for the nuclear dynamics afterwards. First, nuclei' kinetic energy is neglected, as their positions are assumed to be fixed:

$$\sum_{I} \frac{\nabla_{I}^{2}}{2M_{I}} = 0 \quad \text{and} \quad E = E_{\text{tot}} - \sum_{I \le J} \frac{Z_{I}Z_{J}}{|\mathbf{R}_{I} - \mathbf{R}_{J}|}$$

$$(2.16)$$

Following, we define the Coulomb potential of the nuclei experienced by the electrons as:

$$V_{\rm n}(\mathbf{r}) = -\sum_{I} \frac{Z_I}{|\mathbf{r} - \mathbf{R}_I|} \tag{2.17}$$

Then, Equation 2.14 can be rewritten as:

$$\left[ -\sum_{i} \frac{\nabla_{i}^{2}}{2} + \sum_{i} V_{n}(\mathbf{r}_{i}) + \frac{1}{2} \sum_{i \neq j} \frac{1}{|\mathbf{r}_{i} - \mathbf{r}_{j}|} \right] \Psi = E \Psi$$
 (2.18)

Finally, by using Equation 2.15, we can define the electronic and nuclear Schrödinger equations as follows:

$$\left[ -\sum_{i} \frac{\nabla_{i}^{2}}{2} + \sum_{i} V_{\mathbf{n}}(\mathbf{r}_{i}) + \frac{1}{2} \sum_{i \neq j} \frac{1}{|\mathbf{r}_{i} - \mathbf{r}_{j}|} \right] \psi_{\mathbf{R}} = E_{\mathbf{R}} \psi_{\mathbf{R}}$$
(2.19)

$$\left[ -\sum_{I} \frac{\nabla_{I}^{2}}{2M_{I}} + \sum_{I < J} \frac{Z_{I}Z_{J}}{|\mathbf{R}_{I} - \mathbf{R}_{J}|} + E(\mathbf{R}_{1}, \dots, \mathbf{R}_{M}) \right] \chi(\mathbf{R}) = E_{\text{tot}}\chi(\mathbf{R})$$
(2.20)

where  $E_{\mathbf{R}} = E(\mathbf{R}_1, \dots, \mathbf{R}_M)$  is the electronic surface energy, which is a function of the nuclear positions, and serves as an effective potential shaping the nuclear dynamics.

## 2.3 Hartree-Fock Approximation

The essence of the Hartree-Fock approximation (HFA) is to approximate the interacting manyelectron system (Equation 2.18) by a set of non-interacting single-particle problems subject to an effective mean-field potential <sup>13,14,15</sup>. As a means to this, we first rewrite the total wavefunction for a system with N electrons as the product of single-electron wavefunctions—often referred to as the Hartree approximation <sup>16</sup>—as showcased in Equation 2.21.

$$\Psi^{H}(\mathbf{r}_{1},\ldots,\mathbf{r}_{N}) = \prod_{i=1}^{N} \phi_{i}(\mathbf{r}_{i})$$
(2.21)

Following, we construct the total energy functional as the expectation value of the Hamiltonian operator:

$$E^{H}[\{\phi_{i}\}] = \left\langle \Psi^{H} \middle| \hat{H} \middle| \Psi^{H} \right\rangle \tag{2.22}$$

Expanding Equation 2.22:

$$E^{H}[\{\phi_{i}\}] = \sum_{i=1}^{N} \int \phi_{i}^{*}(\mathbf{r}) \left(-\frac{\nabla^{2}}{2} + V_{n}(\mathbf{r})\right) \phi_{i}(\mathbf{r}) d\mathbf{r} + \frac{1}{2} \sum_{i \neq j} \int \int \frac{|\phi_{i}(\mathbf{r})|^{2} |\phi_{j}(\mathbf{r}')|^{2}}{|\mathbf{r} - \mathbf{r}'|} d\mathbf{r} d\mathbf{r}'$$
(2.23)

where the first term adds up the kinetic energy and electron-nuclear attraction overall electrons, whilst the second accounts for the classical electron-electron repulsion energy averaged over the electron density distribution. In order to find the set of orbitals  $\{\phi_i\}$  that minimises the total energy functional, we use the variational principle where we shall impose the orthonormality condition:

$$\int \phi_i^*(\mathbf{r})\phi_j(\mathbf{r})d\mathbf{r} = \delta_{ij} \tag{2.24}$$

for what we introduce the Lagrange multipliers  $\lambda_{ij}$  to enforce these constraints and define the Lagrangian:

$$\mathcal{L} = E^{H}[\{\phi_i\}] - \sum_{i=1}^{N} \lambda_{ij} (\langle \phi_i | \phi_j \rangle - \delta_{ij})$$
(2.25)

which ultimately simplifies to:

$$\mathcal{L} = E^{H}[\{\phi_{i}\}] - \sum_{i=1}^{N} \varepsilon_{i}(\langle \phi_{i} | \phi_{i} \rangle - 1)$$
(2.26)

Then, we need to compute the derivative of  $\mathcal{L}$  with respect to  $\phi_i^*$  and set it to zero:

$$\frac{\delta \mathcal{L}}{\delta \phi_{i}^{*}}(\mathbf{r}) = 0 \tag{2.27}$$

which yields to the Hartree equation:

$$\left[ -\frac{\nabla^2}{2} + V_{\rm n}(\mathbf{r}) + V_i^H(\mathbf{r}) \right] \phi_i(\mathbf{r}) = \varepsilon_i \phi_i(\mathbf{r})$$
 (2.28)

where  $V_n(\mathbf{r})$  represents the electrostatic interaction between electrons and nuclei, and the Hartree potential

$$V_i^H(\mathbf{r}) = \sum_{j \neq i} \int \frac{|\phi_j(\mathbf{r}')|^2}{|\mathbf{r} - \mathbf{r}'|} d\mathbf{r}'$$
(2.29)

accounts for the average electrostatic interaction experienced by the i-th electron due to all other electrons in the system. This effective mean-field potential replaces the electron-electron interactions, effectively simplifying the many-body problem into single-particle problems.

### 2.3.1 The Pauli Exclusion Principle

So far, we have introduced the Hartree approximation, which assumes that the many-electron wavefunction can be expressed as a product of single-particle wavefunctions. However, this approach does not account for the indistinguishability of electrons and the Pauli exclusion principle, which states that no two fermions— half-spin particles, such as electrons—can reside in the same quantum state simultaneously. In doing so, it imposes a restriction on the possible configurations of electrons in a system that shall be accounted for.

In order to achieve this, V. Fock<sup>17</sup> introduced a different approximation to the wavefunction by using a Slater determinant—a mathematical construct that combines one-electron wavefunctions in such a way that satisfies the antisymmetry principle. This is done by expressing the overall wavefunction as the determinant of a matrix of single-electron wavefunctions:

$$\Psi^{HF}(\mathbf{r}_{1},\ldots,\mathbf{r}_{N}) = \frac{1}{\sqrt{N!}} \begin{vmatrix} \phi_{1}(\mathbf{r}_{1}) & \phi_{1}(\mathbf{r}_{2}) & \ldots & \phi_{1}(\mathbf{r}_{N}) \\ \phi_{2}(\mathbf{r}_{1}) & \phi_{2}(\mathbf{r}_{2}) & \ldots & \phi_{2}(\mathbf{r}_{N}) \\ \vdots & \vdots & \ddots & \vdots \\ \phi_{N}(\mathbf{r}_{1}) & \phi_{N}(\mathbf{r}_{2}) & \ldots & \phi_{N}(\mathbf{r}_{N}) \end{vmatrix}$$
(2.30)

where  $1/\sqrt{N!}$  is a normalisation factor. To illustrate this, consider a two-electron system with single-particle wavefunctions  $\phi_1(\mathbf{r})$  and  $\phi_2(\mathbf{r})$ . The Slater determinant for this system would be:

$$\Psi^{HF}(\mathbf{r}_1, \mathbf{r}_2) = \frac{1}{\sqrt{2}} \begin{vmatrix} \phi_1(\mathbf{r}_1) & \phi_1(\mathbf{r}_2) \\ \phi_2(\mathbf{r}_1) & \phi_2(\mathbf{r}_2) \end{vmatrix} = \frac{1}{\sqrt{2}} \left[ \phi_1(\mathbf{r}_1) \phi_2(\mathbf{r}_2) - \phi_1(\mathbf{r}_2) \phi_2(\mathbf{r}_1) \right]$$
(2.31)

Evidently,  $\Psi^{HF}(\mathbf{r}_1, \mathbf{r}_2) = -\Psi^{HF}(\mathbf{r}_2, \mathbf{r}_1)$ , which satisfies the antisymmetr principle.

#### 2.3.2 The Hartree-Fock Equations

The Hartree-Fock equations are derived in a similar manner we addressed the Hartree equations. We first define the total energy with the Hartree-Fock wavefunction (Equation 2.30)

$$E^{HF}[\{\phi_{i}\}] = \left\langle \Psi^{HF} \middle| \hat{H} \middle| \Psi^{HF} \right\rangle$$

$$= \sum_{i} \left\langle \phi_{i} \middle| \frac{\nabla^{2}}{2} + V_{n}(\mathbf{r}) \middle| \phi_{i} \right\rangle$$

$$+ \frac{1}{2} \sum_{i \neq j} \left\langle \phi_{i} \phi_{j} \middle| \frac{1}{|\mathbf{r}_{i} - \mathbf{r}_{j}|} \middle| \phi_{i} \phi_{j} \right\rangle$$

$$- \frac{1}{2} \sum_{i \neq j} \left\langle \phi_{i} \phi_{j} \middle| \frac{1}{|\mathbf{r}_{i} - \mathbf{r}_{j}|} \middle| \phi_{j} \phi_{i} \right\rangle$$

$$(2.32)$$

Consequently, using the variational principle, we derive the Hartree-Fock equations:

$$\left[ -\frac{\nabla^2}{2} + V_{\rm n}(\mathbf{r}) + V_i^H(\mathbf{r}) + \right] \phi_i(\mathbf{r}) - \sum_{j \neq i} \langle \phi_j | \frac{1}{|\mathbf{r}_i - \mathbf{r}_j|} | \phi_i \rangle \phi_j(\mathbf{r}) = \varepsilon_i \phi_i(\mathbf{r})$$
 (2.33)

Noticeably, Equation 2.33 has an extra term compared with the Hartree equation (Equation 2.28). This term is called the "exchange" term<sup>12</sup>, and describes the effects of exchange between electrons. It is convenient to try to express the Hartree-Fock equations in a more compact form, so we define the single-particle and total densities as

$$\rho_i(\mathbf{r}) = |\phi_i(\mathbf{r})|^2 \tag{2.34}$$

$$\rho(\mathbf{r}) = \sum_{i} \rho_i(\mathbf{r}) \tag{2.35}$$

so the Hartree potential can be expressed as

$$V_i^H(\mathbf{r}) = \sum_{j \neq i} \int \frac{\rho_j(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} d\mathbf{r}' = \int \frac{\rho(\mathbf{r}') - \rho_i(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} d\mathbf{r}'$$
(2.36)

Therefore, the single-particle exchange density can be constructed as

$$\rho_i^X(\mathbf{r}, \mathbf{r}') = \sum_{j \neq i} \frac{\phi_i(\mathbf{r}')\phi_i^*(\mathbf{r})\phi_j(\mathbf{r})\phi_j^*(\mathbf{r}')}{\phi_i(\mathbf{r})\phi_i^*(\mathbf{r})}$$
(2.37)

Finally, the Hartree-Fock equations take the form

$$\left[ -\frac{\nabla^2}{2} + V_{\rm n}(\mathbf{r}) + V_i^H(\mathbf{r}) + V_i^X(\mathbf{r}) \right] \phi_i(\mathbf{r}) = \varepsilon_i \phi_i(\mathbf{r})$$
 (2.38)

where  $V_i^X$  stands for the exchange potential

$$V_i^X(\mathbf{r}) = -\int \frac{\rho_i^X(\mathbf{r}, \mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} d\mathbf{r}'$$
(2.39)

## 2.4 Density Functional Theory

So far, we have acknowledged that determining the state of a system with N electrons remains a formidable challenge, for it involves a wavefunction defined in a 3N-dimensional space. We also recognise that it is possible—heuristically speaking—to simplify such representation by utilising products of single-particle wavefunctions. Nevertheless, such independent electrons approximation necessitates the wavefunctions to be explicitly specified, thereby yielding a rather drastic approximation for the behaviour of the system. Thus, it is natural to consider a different

approach to develop the appropriate single-particle framework in an exact manner, onto which approximations can be introduced afterwards.

We hereby introduce the Density Functional Theory (DFT), which draws upon the insight that any property of a system of many electrons can be viewed as a functional of the ground-state density  $n(\mathbf{r})^{18}$  (Equation 2.40)—a scalar function defined over three spatial coordinates.

$$n(\mathbf{r}) = N \int \Psi^*(\mathbf{r}, \dots, \mathbf{r}_N) \Psi(\mathbf{r}, \dots, \mathbf{r}_N) d\mathbf{r}_2 \dots d\mathbf{r}_N$$
 (2.40)

The foundational principles of DFT were established in the original papers by Hohenberg, Kohn and Sham<sup>19,20</sup>, where they present two theorems that establish the theoretical framework of DFT. However, for the purposes of this discussion, we shall base our exposition on explanatory texts<sup>18,10,12,11</sup>.

#### 2.4.1 First Hohenberg-Kohn Theorem

**Theorem 2.1** (First Hohenberg-Kohn Theorem). The ground-state electron density  $n(\mathbf{r})$  uniquely determines the external potential  $V(\mathbf{r})$  and, consequently, the ground-state energy  $E_0$  of a many-electron system.

Proof. Suppose two different external potentials,  $V(\mathbf{r})$  and  $V'(\mathbf{r})$  (different ionic potentials) yield the same ground-state electron density  $n(\mathbf{r})$ . Given that  $V(\mathbf{r})$  and  $V'(\mathbf{r})$  are different in a non-trivial way, we will show that this statement leads to a contradiction. Let E and  $\Psi$  be the total energy and wavefunction and E' and  $\Psi'$  be the total energy and wavefunction corresponding to the systems with hamiltonians  $\hat{H}$  and  $\hat{H}'$ , respectively, with the first hamiltonian containing  $V'(\mathbf{r})$  and the second containing  $V'(\mathbf{r})$  as an external potential:

$$\hat{H} = \hat{T} + \hat{U} + V, \quad \hat{H}' = \hat{T} + \hat{U} + V', \quad E = \langle \Psi | \hat{H} | \Psi \rangle, \quad E' = \langle \Psi' | \hat{H}' | \Psi' \rangle$$

Here,  $\hat{T}$  and  $\hat{U}$  correspond to the kinetic and interaction energy operators, thereby being common for both hamiltonians. Now, we assume that the ground states of the two hamiltonians are different because the external potentials are different. Then, according to the variational principle:

$$E < \langle \Psi' | \hat{H} | \Psi' \rangle = \langle \Psi' | \hat{T} + \hat{U} + V + V' - V' | \Psi' \rangle$$

$$= \langle \Psi' | \hat{H}' + V - V' | \Psi' \rangle$$

$$= E' + \langle \Psi' | (V - V') | \Psi' \rangle$$
(2.41)

Following the same reasoning, we can prove that

$$E' < E - \langle \Psi | (V - V') | \Psi \rangle \tag{2.42}$$

Adding Equations 2.41 and 2.42, we obtain:

$$E + E' < E' + E - \langle \Psi | (V - V') | \Psi \rangle + \langle \Psi' | (V - V') | \Psi' \rangle \tag{2.43}$$

where the last two terms result in

$$\int n'(\mathbf{r})(V - V')d\mathbf{r} - \int n(\mathbf{r})(V - V')d\mathbf{r} = 0$$
(2.44)

since  $n(\mathbf{r}) = n'(\mathbf{r})$  by assumption. Finally, we arrive at the following expression:

$$E + E' < E + E' \tag{2.45}$$

Evidently, this is a contradiction, which implies that our initial assumption about the densities being the same ought to be false, thereby proving there is a one-to-one correspondence between an external potential  $V(\mathbf{r})$  and the electron density  $n(\mathbf{r})$ . Moreover, since  $V(\mathbf{r})$  determines the wavefunction, the wavefunction must be a unique functional of the density. So we conclude that the expression

$$\mathcal{F}[n(\mathbf{r})] = \langle \Psi | \hat{T} + \hat{U} | \Psi \rangle \tag{2.46}$$

must be a universal functional of the electronic density—i.e., common to all solids—and that the ground state energy is a functional of the density:

$$E[n(\mathbf{r})] = \mathcal{F}[n(\mathbf{r})] + \int V(\mathbf{r})n(\mathbf{r})d\mathbf{r}$$
(2.47)

#### 2.4.2 Second Hohenberg-Kohn Theorem

**Theorem 2.2** (Second Hohenberg-Kohn Theorem). The ground-state energy E can be obtained through the variation of trial densities  $\tilde{n}(\mathbf{r})$  instead of trial wavefunctions  $\tilde{\Psi}$ .

*Proof.* First, we fix a trial density  $\tilde{n}(\mathbf{r})$  and define the trial wavefunctions  $\tilde{\Psi}^{\alpha}_{\tilde{n}(\mathbf{r})}$ . Therefore, the constrained energy minimum is defined as

$$E[\tilde{n}(\mathbf{r})] = \min_{\alpha} \langle \tilde{\Psi}^{\alpha}_{\tilde{n}(\mathbf{r})} | \hat{H} | \tilde{\Psi}^{\alpha}_{\tilde{n}(\mathbf{r})} \rangle$$

$$= \mathcal{F}[\tilde{n}(\mathbf{r})] + \int V(\mathbf{r}) \tilde{n}(\mathbf{r}) d\mathbf{r}$$
(2.48)

Secondly, we minimise Equation 2.48 over all n

$$E = \min_{\tilde{n}(\mathbf{r})} E[\tilde{n}(\mathbf{r})] = \min_{\tilde{n}(\mathbf{r})} \left\{ \mathcal{F}[\tilde{n}(\mathbf{r})] + \int V(\mathbf{r})\tilde{n}(\mathbf{r})d\mathbf{r} \right\}$$
(2.49)

For a non-degenerate ground state, the minimum corresponds to the ground-state  $n(\mathbf{r})$ , or to one of the ground-state densities otherwise.

Finally, we have managed to map the formidable challenge of finding the minimum of  $\langle \Psi | \hat{H} | \Psi \rangle$  involving a 3N-dimensional wavefunction into a much simpler problem of finding the minimum of  $E[n(\mathbf{r})]$  involving a 3-dimensional function. This is the essence of DFT, which allows us to compute the ground-state properties of many-electron systems without explicitly solving the many-body Schrödinger equation.

### 2.4.3 Kohn-Sham Equations

Even though the Hohenberg-Kohn theorems provide a rigorous foundation for DFT, they offer no guidance whatsoever for constructing the functional  $\mathcal{F}[n(\mathbf{r})]$ . As such, density functional theory would lack practical utility if it were not for the auxiliary system proposed by Kohn and Sham<sup>18</sup>. It consists of replacing the many-electron problem by an auxiliary independent-particle problem that yields the same ground-state density, incorporating the many-body effects into a so-called exchange-correlation functional. To this end, we first define the density of the auxiliary system as

$$n(\mathbf{r}) = \sum_{i} |\phi_i(\mathbf{r})|^2 \tag{2.50}$$

where  $\phi_i(\mathbf{r})$  are the single-particle wavefunctions of the auxiliary system. Next, we define the independent-particle kinetic energy functional as

$$T_s[n(\mathbf{r})] = \frac{1}{2} \sum_i \int \phi_i^*(\mathbf{r})(-\nabla^2)\phi_i(\mathbf{r})d\mathbf{r}$$
 (2.51)

and the Hartree energy functional can be redefined as

$$E_H[n(\mathbf{r})] = \frac{1}{2} \int \frac{n(\mathbf{r})n(\mathbf{r'})}{|\mathbf{r} - \mathbf{r'}|} d\mathbf{r} d\mathbf{r'}$$
(2.52)

yielding the following expression for the total energy functional:

$$E[n(\mathbf{r})] = T_s[n(\mathbf{r})] + \int V_n(\mathbf{r})n(\mathbf{r})d\mathbf{r} + E_H[n(\mathbf{r})] + E_{xc}[n(\mathbf{r})]$$
(2.53)

where  $E_{xc}[n(\mathbf{r})]$  is the exchange-correlation energy functional is defined as

$$E_{xc}[n(\mathbf{r})] = \langle \hat{T} \rangle - T_s[n(\mathbf{r})] + \langle \hat{U} \rangle - E_H[n(\mathbf{r})]$$
(2.54)

having  $\langle \hat{T} \rangle$  and  $\langle \hat{U} \rangle$  being the exact kinetic energy and electron-electron interaction energy. Finally, we choose a variation in the density to be

$$\delta n(\mathbf{r}) = \delta \phi_i^*(\mathbf{r}) \phi_i(\mathbf{r}) \tag{2.55}$$

along with the following constraint

$$\int \delta n(\mathbf{r}) d\mathbf{r} = \int \delta \phi_i^*(\mathbf{r}) \phi_i(\mathbf{r}) d\mathbf{r} = 0$$
(2.56)

and by applying the Kohn-Sham variational principle, we arrive at the Kohn-Sham equations:

$$\left[ -\frac{\nabla^2}{2} + V_{\text{ext}}(\mathbf{r}) + V_H(\mathbf{r}) + V_{xc}(\mathbf{r}) \right] \phi_i(\mathbf{r}) = \varepsilon_i \phi_i(\mathbf{r})$$
 (2.57)

where  $V_{\text{ext}}(\mathbf{r})$  is the external potential,  $V_H(\mathbf{r})$  is the Hartree potential, and  $V_{xc}(\mathbf{r})$  is the exchange-correlation potential defined as

$$V_{xc}(\mathbf{r}) = \frac{\delta E_{xc}[n(\mathbf{r})]}{\delta n(\mathbf{r})}$$
 (2.58)

Our task now shall focus on constructing appropriate approximations for the exchange-correlation functional  $E_{xc}[n(\mathbf{r})]$ .

### 2.4.4 Exchange-Correlation Functionals

The usefulness of DFT relies entirely on whether reliable approximations for the exchange-correlation functional  $E_{xc}[n(\mathbf{r})]$  can be constructed with sufficient accuracy and computational efficiency. Therefore, we shall now present the most commonly used approximations for  $E_{xc}[n(\mathbf{r})]$ , as well as their strengths and weaknesses.

### 2.4.4.1 Local Density Approximation

The simplest—and remarkably effective—approximation for the exchange-correlation functional is the local-density approximation (LDA)<sup>21</sup>. It approximates the exchange-correlation energy of an inhomogeneous electron system by that of a homogeneous electron gas (HEG) having the same electron density.

$$E_{xc}^{LDA}[n(\mathbf{r})] = \int n(\mathbf{r})\epsilon_{xc}(n(\mathbf{r}))d\mathbf{r}$$
 (2.59)

where  $\epsilon_{xc}(n)$  is the exchange and correlation energy per electron of a uniform electron gas of density  $n^{19,20}$ . This quantity depends solely on the local density and surrounding electrons in the vicinity of  $\mathbf{r}$ , for example, a sphere of radius  $\sim \lambda_F(\mathbf{r})$ —the local Fermi wavelength<sup>15</sup>  $\lambda_F(\mathbf{r}) \equiv [3\pi^2 n(\mathbf{r})]^{-1/3}$ .

The exchange and correlation contributions to  $E_{xc}^{LDA}$  can be separated into two terms

$$E_{xc}^{LDA}[n(\mathbf{r})] = E_x^{HEG}[n(\mathbf{r})] + E_c^{HEG}[n(\mathbf{r})]$$
(2.60)

The first term corresponds to the exchange energy density contribution

$$\epsilon_x^{HEG}(n) = -\frac{3}{4} \left(\frac{3}{\pi}\right)^{1/3} n^{1/3}$$

$$= -\frac{0.458128}{r_s}$$
(2.61)

where  $r_s$  is the Wigner-Seitz radius—the radius of a sphere containing one electron and given by  $(4\pi/3)r_s^3 = n^{-1}$ .

The second term corresponds to the correlation energy, which was computed by Ceperley and Alder<sup>22</sup> using quantum Monte Carlo methods. Subsequently, the extracted data was parameterised by Perdew and Zunger<sup>23</sup>, yielding the following expression:

$$\epsilon_c^{HEG}(r_s) = \begin{cases} 0.0311 \ln r_s - 0.0480 + 0.002r_s \ln r_s - 0.0116r_s & r_s < 1\\ -0.1423 & r_s \ge 1 \end{cases}$$
(2.62)

LDA has demonstrated remarkable success for most applications involving systems with slowly varying densities or systems with high densities. Nevertheless, LDA—and its spin-polarised version (LSDA)— breaks down in systems governed by strong correlation effects that it loses any resemblance to non-interacting electron gases.

#### 2.4.4.2 Generalised Gradient Approximation

In contrast to LDA—which considers the electronic density to be locally uniform—the generalised gradient approximation (GGA) systematically improves upon LDA by incorporating not only the local density  $n(\mathbf{r})$ , but also its gradient  $\nabla n(\mathbf{r})$ , thereby accounting for the inhomogeneities in the electron distribution.

Within this framework, the exchange-correlation energy functional is expressed as a function of the local density and its gradient

$$E_{xc}^{GGA}[n(\mathbf{r})] = \int n(\mathbf{r}) \epsilon_{xc}^{HEG}(n(\mathbf{r})) f(n_{\uparrow}(\mathbf{r}), n_{\downarrow}(\mathbf{r}), \nabla n_{\uparrow}(\mathbf{r}), \nabla n_{\downarrow}(\mathbf{r})) d\mathbf{r}$$
(2.63)

where  $n_{\uparrow}(\mathbf{r})$  and  $n_{\downarrow}(\mathbf{r})$  are the spin-up and spin-down electron densities, respectively. Additionally, the exact form of f—a parametrised analytic function—depends on the GGA under consideration.

In this regard, one of the most prominent and widely adopted GGA functionals is the Perdew-Burke-Ernzerhof (PBE)<sup>24</sup> functional, which was proposed as a solution for the drawbacks of previously proposed GGAs, such as the Perdew-Wang (PW91) functional. Within PBE, all parameters—other than those in  $\epsilon_{xc}^{HEG}(n(\mathbf{r}))$ —are fundamental constants. Consequently, the exchange energy term in the PBE functional is given by

$$E_x^{PBE} = \int n(\mathbf{r}) \epsilon_x^{HEG}(n(\mathbf{r})) \left[ 1 + \kappa - \frac{\kappa}{1 + \mu s^2 / \kappa} \right] d\mathbf{r}$$
 (2.64)

where  $\kappa = 0.804$  and  $\mu = 0.219$  are the parameters of the PBE functional,  $k_F = (3\pi^2 n(\mathbf{r}))^{1/3}$  is the local Fermi wavevector and  $s = |\nabla n(\mathbf{r})|/(2k_F n(\mathbf{r}))$  is a dimensionless density gradient.

The correlation energy term in the PBE functional is expressed as

$$E_c^{PBE} = \int n(\mathbf{r}) \left[ \epsilon_c^{HEG} + \gamma \phi^3 \ln \left\{ 1 + \frac{\beta}{\gamma} t^2 \left[ \frac{1 + At^2}{1 + At^2 + A^2 t^4} \right] \right\} \right]$$
 (2.65)

where  $\gamma = 0.031091, \, \beta = 0.066725, \, \phi$  is a spin-scaling factor, and A and t are defined as

$$A = \frac{\beta}{\gamma} \left[ \exp \left\{ \frac{-\epsilon_c^{HEG}}{\gamma \phi^3} \right\} - 1 \right]^{-1}, \quad t(\mathbf{r}) = \frac{|\nabla n(\mathbf{r})|}{2\phi k_s n(\mathbf{r})}$$
 (2.66)

where  $k_s = \sqrt{4k_F/\pi}$  is the Thomas-Fermi screening wavenumber.

Even though PBE holds a significant advantage over LDA in terms of accuracy, it is not exempt from certain limitations and shortcomings. PBE tends to overestimate equilibrium lattice constants by about 1%—LDA underestimates them by the same amount—which is detrimental for accurate calculations of other equilibrium properties, such as bulk moduli, phonon frequencies and magnetism.

To address this issue, a revised version of the PBE—the PBEsol functional<sup>25</sup>—was developed, which improves equilibrium properties of densely-packed solids and their surfaces, reducing the overestimation of lattice constants by a factor of  $\sim 4$ . Nonetheless, PBEsol does not perform well for semiconductors and can lead to less accurate total energy calculations compared to PBE. Ultimately, the choice of GGA functional depends on the specific system and properties being investigated, as well as the desired balance between accuracy and computational efficiency.

#### 2.4.4.3 Hybrid Functionals

Functionals mentioned above and their inherent limitations motivated the exploration of hybrid functionals. They offer improved accuracy by incorporating a fraction of the exact nonlocal Hartree-Fock exchange energy into the exchange-correlation functional, allowing efficient yet accurate calculations.

One of these functionals includes PBE0<sup>26</sup>—a combination of the PBE functional and the exact Hartree-Fock exchange energy. It is defined as

$$E_{xc}^{PBE0} = \frac{1}{4}E_x^{HF} + \frac{3}{4}E_x^{PBE} + E_c^{PBE}$$
 (2.67)

Another prominent example is the HSE (Heyd-Scuseria-Ernzerhof) functional<sup>27</sup>, which splits the exchange energy into short-range (SR) and long-range (LR) contributions

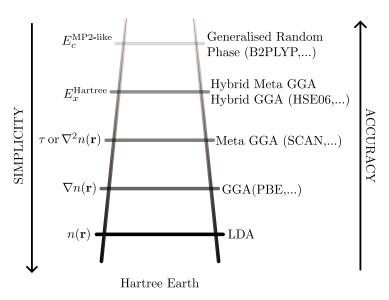
$$E_{xc}^{HSE06} = \frac{1}{4} E_x^{HF,SR}(\omega) + \frac{3}{4} E_x^{PBE,SR}(\omega) + E_c^{PBE,LR}(\omega) + E_c^{PBE}(\omega)$$
 (2.68)

where  $\omega$  is an adjustable parameter that controls the short and long-range separation in the decomposed Coulomb operator

$$\frac{1}{r} = SR_{\omega}(\mathbf{r}) + LR_{\omega}(\mathbf{r}) = \frac{erfc(\omega r)}{r} + \frac{erf(\omega r)}{r}$$
(2.69)

Ultimately, hybrid functionals provide a considerable improvement in the accuracy of electronic structure calculations over LDA and GGA. Nonetheless, this advantage comes at a higher computational cost. This hierarchy of exchange-correlation functionals is better summarised in the so-called Jacob's ladder<sup>28</sup> (Figure 2.1), depicting the trade-off between accuracy and computational cost as we ascend the ladder towards more sophisticated functionals.

#### Chemical Accuracy



**Figure 2.1:** Jacob's ladder of exchange-correlation functionals as proposed by J. Perdew<sup>28</sup>. The ladder categorises the functionals into rungs, from the simplest approximation (LDA) at the bottom progressing to more sophisticated and accurate approximations (Generalised Random Phase) at the uppermost rung.

## 2.5 Density Functional Tight Binding

This is a placeholder

## 2.6 Ab initio Molecular Dynamics

Molecular dynamics (MD) is a widely used technique in condensed matter physics, for it provides a powerful framework to simulate the time evolution of many-body systems and to investigate macroscopic properties of materials, such as temperature, pressure, and energy distributions<sup>29,30</sup>.

Molecular dynamics (MD) is a powerful computational technique serving to investigate many-body condensed matter systems by simulating the time evolution of its constituent particles under the influence of interatomic interactions<sup>29,30</sup>. Given a set of initial conditions—positions and velocities of all atoms in the system, and the interatomic potential—MD provides a means to compute the trajectories of atoms over a time interval by solving Newton's equations of motion.

- 2.6.1 Born-Oppenheimer Molecular Dynamics
- 2.6.2 Car-Parrinello Molecular Dynamics
- 2.7 Computational Implementation in VASP
- 2.7.1 Pseudopotentials
- 2.7.2 K-Point Sampling and Cutoff Energy
- 2.7.3 Projector Augmented Wave Method
- 2.7.4 Machine Learning Interatomic Potentials

## Chapter 3

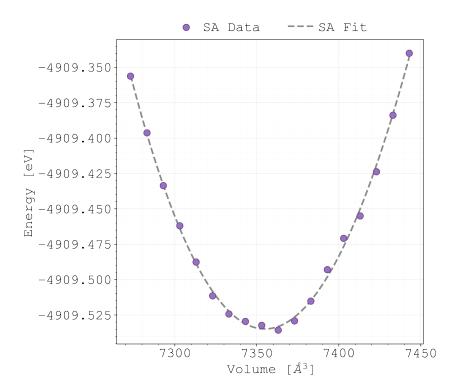
## Methodology

## Chapter 4

## Results & Discussion

## 4.1 My results

Hello



**Figure 4.1:** A schematic representation of the DFT formalism. The many-body wavefunction is replaced by a single-particle wavefunction, which is used to calculate the electron density. The electron density is then used to calculate the total energy of the system. Adapted from  $^{10}$ .

## Bibliography

- [1] Paulo J. M. Monteiro, Sabbie A. Miller, and Arpad Horvath. "Towards Sustainable Concrete". In: Nature Materials 16.7 (July 2017), pp. 698–699. ISSN: 1476-4660. DOI: 10.1038/nmat4930. (Visited on 03/16/2025).
- [2] Henri Van Damme. "Concrete Material Science: Past, Present, and Future Innovations". In: Cement and Concrete Research. SI: Digital Concrete 2018 112 (Oct. 2018), pp. 5–24. ISSN: 0008-8846. DOI: 10.1016/j.cemconres.2018.05.002. (Visited on 03/16/2025).
- [3] P. Kumar Mehta and Paulo J. M. Monteiro. "Introduction". In: *Concrete: Microstructure*, *Properties, and Materials*. 4th Edition. New York: McGraw-Hill Education, 2014. ISBN: 978-0-07-179787-0.
- [4] Styliani Papatzani, Kevin Paine, and Juliana Calabria-Holley. "A Comprehensive Review of the Models on the Nanostructure of Calcium Silicate Hydrates". In: *Construction and Building Materials* 74 (Jan. 2015), pp. 219–234. ISSN: 0950-0618. DOI: 10.1016/j.conbuildmat.2014.10.029. (Visited on 03/16/2025).
- [5] Katerina Ioannidou et al. "Mesoscale Texture of Cement Hydrates". In: Proceedings of the National Academy of Sciences 113.8 (Feb. 2016), pp. 2029–2034. DOI: 10.1073/pnas. 1520487113. (Visited on 01/16/2025).
- [6] Mohammad Javad Abdolhosseini Qomi, Mathieu Bauchy, and Roland J.-M. Pellenq. "Nanoscale Composition-Texture-Property Relation in Calcium-Silicate-Hydrates". In: *Handbook of Materials Modeling: Applications: Current and Emerging Materials*. Ed. by Wanda Andreoni and Sidney Yip. Cham: Springer International Publishing, 2020, pp. 1761–1792. ISBN: 978-3-319-44680-6. DOI: 10.1007/978-3-319-44680-6\_128. (Visited on 09/24/2024).
- [7] Ashraf A. Bahraq et al. "Molecular Simulation of Cement-Based Materials and Their Properties". In: *Engineering* 15 (Aug. 2022), pp. 165–178. ISSN: 2095-8099. DOI: 10.1016/j.eng.2021.06.023. (Visited on 09/17/2024).
- [8] Emmy M. Foley, Jung J. Kim, and M. M. Reda Taha. "Synthesis and Nano-Mechanical Characterization of Calcium-Silicate-Hydrate (C-S-H) Made with 1.5 CaO/SiO2 Mixture". In: Cement and Concrete Research 42.9 (Sept. 2012), pp. 1225–1232. ISSN: 0008-8846. DOI: 10.1016/j.cemconres.2012.05.014. (Visited on 06/05/2025).

Bibliography 23

[9] Mohammad Javad Abdolhosseini Qomi, Franz-Josef Ulm, and Roland J.-M. Pellenq. "Physical Origins of Thermal Properties of Cement Paste". In: *Physical Review Applied* 3.6 (June 2015), p. 064010. DOI: 10.1103/PhysRevApplied.3.064010. (Visited on 09/11/2024).

- [10] F. Giustino. Materials Modelling Using Density Functional Theory: Properties and Predictions. Oxford University Press, 2014. ISBN: 978-0-19-966244-9.
- [11] D.S. Sholl and J.A. Steckel. *Density Functional Theory: A Practical Introduction*. Wiley, 2023. ISBN: 978-1-119-84086-2.
- [12] E. Kaxiras. Atomic and Electronic Structure of Solids. Cambridge University Press, 2003. ISBN: 978-0-521-81010-4.
- [13] R.M. Martin, L. Reining, and D.M. Ceperley. *Interacting Electrons*. Cambridge University Press, 2016. ISBN: 978-0-521-87150-1.
- [14] T. Helgaker, P. Jorgensen, and J. Olsen. Molecular Electronic-Structure Theory. Wiley, 2014. ISBN: 978-1-119-01955-8.
- [15] D. Feng and G. Jin. *Introduction to Condensed Matter Physics, Volume 1.* World Scientific Publishing Company, 2005. ISBN: 978-981-310-219-4.
- [16] D. R. Hartree. "The Wave Mechanics of an Atom with a Non-Coulomb Central Field. Part I. Theory and Methods". In: Mathematical Proceedings of the Cambridge Philosophical Society 24.1 (Jan. 1928), pp. 89–110. ISSN: 1469-8064, 0305-0041. DOI: 10.1017/S0305004100011919. (Visited on 07/10/2025).
- [17] V. Fock. "Näherungsmethode zur Lösung des quantenmechanischen Mehrkörperproblems".
   In: Zeitschrift für Physik 61.1 (Jan. 1930), pp. 126–148. ISSN: 0044-3328. DOI: 10.1007/BF01340294. (Visited on 07/10/2025).
- [18] R.M. Martin. Electronic Structure: Basic Theory and Practical Methods. Cambridge University Press, 2020. ISBN: 978-1-108-42990-0.
- [19] P. Hohenberg and W. Kohn. "Inhomogeneous Electron Gas". In: *Physical Review* 136.3B
   (Nov. 1964), B864–B871. DOI: 10.1103/PhysRev.136.B864. (Visited on 07/12/2025).
- [20] W. Kohn and L. J. Sham. "Self-Consistent Equations Including Exchange and Correlation Effects". In: *Physical Review* 140.4A (Nov. 1965), A1133–A1138. DOI: 10.1103/PhysRev. 140.A1133. (Visited on 07/12/2025).
- [21] W. Kohn. "Nobel Lecture: Electronic Structure of Matter—Wave Functions and Density Functionals". In: *Reviews of Modern Physics* 71.5 (1999), pp. 1253–1266. DOI: 10.1103/RevModPhys.71.1253.
- [22] D. M. Ceperley and B. J. Alder. "Ground State of the Electron Gas by a Stochastic Method".
   In: Physical Review Letters 45.7 (Aug. 1980), pp. 566-569. DOI: 10.1103/PhysRevLett.45.
   566. (Visited on 07/19/2025).

Bibliography 24

[23] J. P. Perdew and Alex Zunger. "Self-Interaction Correction to Density-Functional Approximations for Many-Electron Systems". In: *Physical Review B* 23.10 (May 1981), pp. 5048–5079. DOI: 10.1103/PhysRevB.23.5048. (Visited on 07/19/2025).

- [24] John P. Perdew, Kieron Burke, and Matthias Ernzerhof. "Generalized Gradient Approximation Made Simple". In: *Physical Review Letters* 77.18 (Oct. 1996), pp. 3865–3868. DOI: 10.1103/PhysRevLett.77.3865. (Visited on 07/19/2025).
- [25] John P. Perdew et al. "Restoring the Density-Gradient Expansion for Exchange in Solids and Surfaces". In: *Physical Review Letters* 100.13 (Apr. 2008), p. 136406. DOI: 10.1103/PhysRevLett.100.136406. (Visited on 07/21/2025).
- [26] Jochen Heyd, Gustavo E. Scuseria, and Matthias Ernzerhof. "Hybrid Functionals Based on a Screened Coulomb Potential". In: The Journal of Chemical Physics 118.18 (May 2003), pp. 8207–8215. ISSN: 0021-9606. DOI: 10.1063/1.1564060. (Visited on 07/19/2025).
- [27] Jonathan E. Moussa, Peter A. Schultz, and James R. Chelikowsky. "Analysis of the Heyd-Scuseria-Ernzerhof Density Functional Parameter Space". In: *The Journal of Chemical Physics* 136.20 (May 2012). ISSN: 0021-9606, 1089-7690. DOI: 10.1063/1.4722993. arXiv: 1205.4999 [cond-mat]. (Visited on 07/22/2025).
- [28] John P. Perdew and Karla Schmidt. "Jacob's Ladder of Density Functional Approximations for the Exchange-Correlation Energy". In: AIP Conference Proceedings 577.1 (July 2001), pp. 1–20. ISSN: 0094-243X. DOI: 10.1063/1.1390175. (Visited on 07/19/2025).
- [29] D. Marx and J. Hutter. Ab Initio Molecular Dynamics: Basic Theory and Advanced Methods. Cambridge University Press, 2009. ISBN: 978-1-139-47719-2.
- [30] D. Frenkel and B. Smit. Understanding Molecular Simulation: From Algorithms to Applications. Academic Press, 2023. ISBN: 978-0-323-91318-8.