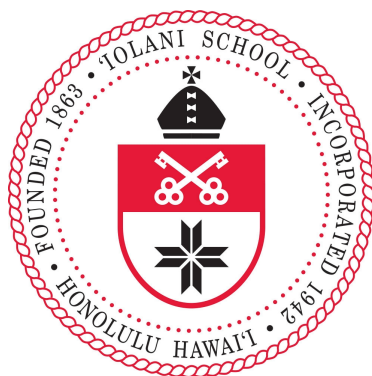


'Iolani Science Olympiad Pre-Nationals Practice

Division C



MATERIALS SCIENCE TEST



Name _____

Honolulu
2025

1 The Big Bad Multiple Choice Questions

1. In a face-centered cubic (FCC) lattice of ions, what fraction of the total lattice energy of the solid arises from interactions beyond the first coordination shell, assuming a point-charge model and using the Madelung constant?
(A) $\sim 5\%$
(B) $\sim 15\%$
(C) $\sim 30\%$
(D) $\sim 50\%$
2. Which of the following Bravais lattices cannot produce a powder X-ray diffraction pattern with systematically absent reflections of the form $h + k + l = \text{odd}$?
(A) Body-centered cubic (BCC)
(B) Face-centered cubic (FCC)
(C) Simple cubic (SC)
(D) Hexagonal close-packed (HCP)
3. The theoretical maximum packing efficiency of the hexagonal close-packed structure is:
(A) 68.0%
(B) 74.0%
(C) 52.4%
(D) 80.0%
4. In the rock-salt (NaCl) structure, the radius of the largest octahedral void (in units of the host sphere radius R) is:
(A) $0.155 R$
(B) $0.414 R$
(C) $0.732 R$
(D) $1.000 R$
5. Which of these oxide glass formers exhibits the highest glass transition temperature (T_g)?
(A) SiO_2
(B) B_2O_3
(C) P_2O_5
(D) GeO_2
6. In silicate glasses, increasing the fraction of modifier cations (e.g. Na^+) primarily:
(A) Increases network connectivity and raises T_g

- (B) Breaks Si–O–Si linkages and lowers T_g
 - (C) Converts bridging oxygens to non-bridging oxygens but raises density
 - (D) Has negligible effect on optical refractive index
7. According to Zachariasen’s rules for glass formation, which is *not* one of the rules?
- (A) Oxygen atoms are linked to no more than two glass-forming atoms.
 - (B) Each glass-forming atom must be surrounded by a coordination number of at least three.
 - (C) Polyhedra share corners, not edges or faces.
 - (D) A random three-dimensional network is formed.
8. The so-called “fragility” of a glass-forming liquid describes:
- (A) The steepness of the viscosity vs. temperature plot at T_g
 - (B) The susceptibility of the glass to mechanical fracture
 - (C) The amount of non-bridging oxygens in the network
 - (D) The difference between T_g and crystallization temperature T_x
9. Phase separation in alkali silicate glasses upon cooling is often a result of:
- (A) Immiscibility between silica-rich and alkali-rich regions
 - (B) Crystallization of quartz nuclei
 - (C) Spinodal decomposition of polymer chains
 - (D) Diffusion-limited aggregation of modifier ions
10. Which pair of elements would exhibit the greatest degree of covalent character according to Pauling’s scale of electronegativity?
- (A) Na–Cl
 - (B) Si–O
 - (C) Mg–F
 - (D) Al–N
11. The Born exponent n in the Born–Mayer potential is most directly related to:
- (A) The compressibility of the crystal
 - (B) The ionic radii ratio
 - (C) The polarizability of the anion
 - (D) The bond dissociation energy at zero separation
12. Which of the following materials would you expect to have the highest cohesive energy?
- (A) Metallic sodium
 - (B) Diamond (carbon)

- (C) Ionic sodium chloride
 - (D) Amorphous silica glass
13. The work function of a metal is primarily governed by:
- (A) Van der Waals forces between atoms
 - (B) Ionization energy of the isolated atom
 - (C) Strength of metallic bonding and electron density at the surface
 - (D) Electron affinity of the metal atom
14. Madelung's constant for the wurtzite (hexagonal ZnS) lattice is about 1.64. Which of these statements follows?
- (A) The lattice energy of ZnS is lower than that of NaCl (Madelung ≈ 1.747) for the same interionic distance.
 - (B) ZnS must be metallic, not ionic.
 - (C) The coordination number in wurtzite is greater than in rock salt.
 - (D) Electrostatic energy is independent of Madelung constant.
15. The minimum radius ratio $r_{\text{cation}}/r_{\text{anion}}$ required for a cation to occupy a tetrahedral site in an anion lattice is approximately:
- (A) 0.155
 - (B) 0.225
 - (C) 0.414
 - (D) 0.732
16. A compound with $r_{\text{cat}}/r_{\text{an}} \approx 0.732$ is most likely to adopt which coordination geometry?
- (A) Linear (CN = 2)
 - (B) Tetrahedral (CN = 4)
 - (C) Octahedral (CN = 6)
 - (D) Cubic (CN = 8)
17. Which of these Laves phases (AB_2) is favored when $r_A/r_B \approx 1.225$?
- (A) MgCu_2 (C15, cubic)
 - (B) MgZn_2 (C14, hexagonal)
 - (C) MgNi_2 (C36, hexagonal)
 - (D) Any of the above; Laves phases are insensitive to radius ratios
18. In a mixed halide perovskite, if the average anion radius increases by 10%, the perovskite tolerance factor t will:
- (A) Increase by $\sim 10\%$
 - (B) Decrease by $\sim 5\%$
 - (C) Stay constant

- (D) Increase, but by less than 10%
- 19.** For alkali halides, the transition from the rock-salt to the wurtzite/ZnS structure occurs when $r_{\text{cat}}/r_{\text{an}} \lesssim 0.414$. Which of these should therefore show a wurtzite-type structure under ambient conditions?
- (A) LiF ($r^+ = 0.76 \text{ \AA}$; $r^- = 1.33 \text{ \AA}$)
 (B) NaCl ($r^+ = 1.02 \text{ \AA}$; $r^- = 1.81 \text{ \AA}$)
 (C) KBr ($r^+ = 1.38 \text{ \AA}$; $r^- = 1.96 \text{ \AA}$)
 (D) CsI ($r^+ = 1.67 \text{ \AA}$; $r^- = 2.20 \text{ \AA}$)
- 20.** The energy of a small-angle tilt boundary varies approximately as:
- $$\gamma \propto Gb\theta$$
- where G is the shear modulus, b the Burgers vector, and θ the misorientation angle (in radians). This relationship holds for θ up to about:
- (A) 1°
 (B) 5°
 (C) 10°
 (D) 15°
- 21.** A $\Sigma 3$ coincidence-site lattice (CSL) boundary in a cubic metal corresponds to a misorientation of:
- (A) 36.9° about $\langle 110 \rangle$
 (B) 60° about $\langle 111 \rangle$
 (C) 38.9° about $\langle 111 \rangle$
 (D) 43.3° about $\langle 100 \rangle$
- 22.** Which mechanism predominantly lowers the energy of a high-angle grain boundary upon annealing?
- (A) Boundary migration reducing total boundary area
 (B) Vacancy diffusion within the boundary core
 (C) Precipitate pinning at the boundary
 (D) Grain rotation without migration
- 23.** Which imperfection introduces tensile stress fields in the surrounding lattice?
- (A) Interstitial atom
 (B) Vacancy
 (C) Edge dislocation extra half-plane
 (D) Screw dislocation
- 24.** If the vacancy formation energy doubles, at the same temperature the equilibrium concentration of vacancies:
- (A) Doubles

- (B) Halves
 - (C) Squares
 - (D) Becomes its own square root
- 25.** In a metal at $0.5 T_m$ (half melting temperature), the vacancy concentration is 10^{-4} . At $0.6 T_m$, roughly what is the concentration, assuming constant vacancy formation energy?
- (A) 10^{-6}
 - (B) 10^{-5}
 - (C) 10^{-3}
 - (D) 10^{-2}
- 26.** For a cation of radius ratio $r_c/r_a = 0.30$ in an anion FCC lattice, the preferred site is:
- (A) Tetrahedral
 - (B) Octahedral
 - (C) Cubic
 - (D) None (unstable)
- 27.** Segregation of impurity atoms to grain boundaries lowers the boundary energy because of:
- (A) Reduction of lattice parameter mismatch
 - (B) Elastic interaction between solute and boundary
 - (C) Increased boundary mobility
 - (D) Precipitation strengthening
- 28.** A typical Frenkel defect in AgCl involves:
- (A) Cl vacancy + Cl interstitial
 - (B) Ag vacancy + Ag interstitial
 - (C) Cl vacancy + Ag interstitial
 - (D) Ag vacancy + Cl interstitial
- 29.** In an ideal Schottky defect in NaCl, the ratio of vacancies is:
- (A) $1 \text{ Na}^+ : 1 \text{ Cl}^-$
 - (B) $2 \text{ Na}^+ : 1 \text{ Cl}^-$
 - (C) $1 \text{ Na}^+ : 2 \text{ Cl}^-$
 - (D) $2 \text{ Na}^+ : 2 \text{ Cl}^-$
- 30.** The activation energy for self-diffusion in ionic crystals often includes both migration and formation energies. If $E_m = 0.5 \text{ eV}$ and $E_f = 1.0 \text{ eV}$, the diffusion activation is:
- (A) 0.5 eV

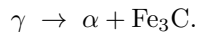
- (B) 1.0 eV
 (C) 1.5 eV
 (D) 2.0 eV
- 31.** The Burgers vector \mathbf{b} of a perfect dislocation in a BCC metal is:
- (A) $a\langle 100 \rangle$
 (B) $\frac{a}{2}\langle 111 \rangle$
 (C) $a\langle 110 \rangle$
 (D) $\frac{a}{2}\langle 110 \rangle$
- 32.** A mixed dislocation with edge and screw character lies at a tilt grain boundary. The edge component drives:
- (A) Boundary climb
 (B) Boundary migration
 (C) Dislocation cross-slip
 (D) Twinning
- 33.** Which compound is least likely to exhibit Frenkel disorder?
- (A) AgBr
 (B) ZnS
 (C) NaCl
 (D) CsI
- 34.** The equilibrium Schottky defect concentration in MgO (one Mg^{2+} + one O^{2-}) varies as:
- $$C_s \propto \exp\left(-\frac{E_f}{2kT}\right).$$
- The factor 1/2 arises because:
- (A) Two ions removed per defect pair
 (B) Charge neutrality constraint
 (C) Migration barrier halved
 (D) Vacancy formation enthalpy split
- 35.** In olivine-type Mg_2SiO_4 , the minor Fe^{2+} occupies:
- (A) Octahedral M1 only
 (B) Octahedral M2 only
 (C) Both M1 and M2
 (D) Tetrahedral sites
- 36.** The effect of hydrostatic pressure on vacancy formation energy is to:
- (A) Increase if defect volume positive
 (B) Decrease if defect volume positive

- (C) No change
 - (D) Reverse defect type
- 37.** The Debye-Waller factor in diffraction includes contribution from point defects via:
- (A) Static displacements only
 - (B) Thermal vibrations only
 - (C) Both static and thermal
 - (D) Neither
- 38.** The Read-Shockley model fails for angles above:
- (A) 5°
 - (B) 15°
 - (C) 30°
 - (D) 45°
- 39.** Substitutional impurity diffusion in fcc metals occurs via:
- (A) Direct interstitial jumps
 - (B) Vacancy mechanism
 - (C) Interstitialcy mechanism
 - (D) Grain boundary only
- 40.** The concentration ratio of Frenkel defects in AgCl at equilibrium is:
- $$\frac{n_{\text{vac}}}{n_{\text{int}}} = ?$$
- Assuming equal formation energy for vacancy and interstitial, the ratio is:
- (A) 1
 - (B) 2
 - (C) Depends on site multiplicity
 - (D) Zero
- 41.** In UO_2 , Schottky defects form more readily than Frenkel because:
- (A) U^{4+} large cation
 - (B) O^{2-} small anion
 - (C) Charge vs. size considerations
 - (D) High oxygen mobility
- 42.** Dislocation climb rate is controlled by:
- (A) Vacancy diffusion
 - (B) Interstitial diffusion
 - (C) Glide plane friction

- (D) Peierls barrier
- 43. The Kirkendall effect demonstrates:
 - (A) Vacancy flow markers
 - (B) Dislocation motion
 - (C) Grain boundary migration
 - (D) Precipitate growth
- 44. In semiconductor doped Si, vacancy concentration under n-type doping:
 - (A) Increases
 - (B) Decreases
 - (C) Unchanged
 - (D) Oscillates
- 45. Hydrogen in metals occupies:
 - (A) Octahedral only
 - (B) Tetrahedral only
 - (C) Both depending on T
 - (D) Substitutional only
- 46. Diffusion along grain boundaries is typically:
 - (A) Slower than lattice
 - (B) Same as lattice
 - (C) Faster than lattice
 - (D) Zero
- 47. The Cottrell atmosphere forms around:
 - (A) Grain boundaries
 - (B) Dislocations
 - (C) Twin boundaries
 - (D) Dr. Cottrell's office
- 48. Which defect pair conserves both mass and charge?
 - (A) Frenkel
 - (B) Schottky
 - (C) Both
 - (D) Neither
- 49. At the invariant eutectic point in the Cu–Sn system (227 °C), how many phases coexist?
 - (A) One
 - (B) Two

- (C) Three
- (D) Four

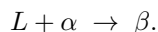
50. In the Fe–Fe₃C phase diagram, the eutectoid transformation at 727 °C is:



This reaction is classified as:

- (A) Eutectic
- (B) Eutectoid
- (C) Peritectic
- (D) Peritectoid

51. A peritectic reaction in a binary system is of the form:



Which of the following is a peritectic system?

- (A) Cu–Ni (complete solid solution)
- (B) Pb–Sn (eutectic)
- (C) Fe–C (eutectoid)
- (D) Cd–Mg (peritectic)

52. The degree of freedom F at an invariant point in a binary isobaric phase diagram is given by Gibbs phase rule. At a eutectic ($P = 3$, $C = 2$), $F =$

- (A) 0
- (B) 1
- (C) 2
- (D) 3

53. In a monotectic reaction ($L_1 \rightarrow L_2 + \alpha$), the phase that exsolves is:

- (A) α
- (B) L_1
- (C) L_2
- (D) Alloys do not show monotectics

54. A congruently melting intermetallic forms a straight vertical liquidus line. Which of these compounds is congruent?

- (A) Au–Cu₃
- (B) Zn–Mg₂
- (C) Ni–Al (NiAl)
- (D) Ag–Cd

55. For a binary isomorphous system, the solidus and liquidus lines coincide at:

- (A) Pure A and pure B only
 - (B) Eutectic composition
 - (C) Peritectic composition
 - (D) Every composition
- 56.** Inside a miscibility gap, the limit of phase separation is defined by the spinodal curve where:
- $$\frac{\partial^2 G}{\partial x^2} = 0.$$
- Between the binodal and spinodal lines, decomposition occurs by:
- (A) Spinodal decomposition
 - (B) Nucleation and growth
 - (C) Martensitic shear
 - (D) Diffusionless transformation
- 57.** In a ternary phase diagram at a fixed temperature, the tie-triangle connects three phases α , β , γ . The overall alloy composition lies inside this triangle; phase fractions are given by:
- (A) Lever rule in 2D via areas
 - (B) Common tangent construction
 - (C) Lever rule on each edge only
 - (D) Cannot be determined
- 58.** In the Al–Si eutectic system, primary α (Al-rich) plates form in a hypoeutectic alloy. If cooling is too fast, primary α may:
- (A) Fully transform to eutectic mixture
 - (B) Become dendritic instead of equiaxed
 - (C) Dissolve back into liquid
 - (D) Form martensite
- 59.** The slope of the liquidus line in an isomorphous binary solid solution follows the Clapeyron relation. If ΔH_{fus} increases, the liquidus slope:
- (A) Increases
 - (B) Decreases
 - (C) Unchanged
 - (D) Becomes negative
- 60.** A peritectoid reaction ($\alpha + \beta \rightarrow \gamma$) occurs entirely in the solid state. Which system shows this?
- (A) Fe–Fe₃C
 - (B) Cu–Ni
 - (C) Ni–Ti
 - (D) Ag–Pd

61. For a binary system with positive heats of mixing, the high-temperature phase diagram exhibits:
- (A) Complete solubility at all T
 - (B) Miscibility gap at high T
 - (C) Monotectic reaction only
 - (D) Spinodal decomposition only
62. In a binary diagram, the eutectic microstructure's lamellar spacing λ scales with the growth rate V as:

$$\lambda^2 V = \text{constant}.$$

If V doubles, λ changes by factor:

- (A) $1/\sqrt{2}$
 - (B) $1/2$
 - (C) $\sqrt{2}$
 - (D) 2
63. In the Fe–C system, pearlite forms by diffusion-controlled transformation along the TTT curve. The nose of the curve corresponds to:
- (A) Maximum transformation rate
 - (B) Start of martensite formation
 - (C) Curie temperature
 - (D) Eutectic point
64. A metastable extension of the liquidus (“metastable eutectic”) appears if kinetic inhibition prevents:
- (A) Nucleation of the stable phase
 - (B) Diffusion in the solid
 - (C) Grain boundary migration
 - (D) Surface oxidation
65. The Gibbs–Thomson effect causes small precipitates to dissolve at temperatures higher than bulk equilibrium. As particle radius r decreases, the dissolution temperature:
- (A) Increases
 - (B) Decreases
 - (C) Unchanged
 - (D) Oscillates
66. In a binary eutectic with limited solubility in solid phases, the lever rule fails near the eutectic because:
- (A) Phase compositions vary with temperature

- (B) Tie lines do not exist
 - (C) Diffusion is too fast
 - (D) Interface energies dominate
- 67.** The “nose” of a TTT diagram corresponds to the minimum time for 50% transformation. For a given alloy, shifting the curve to longer times indicates:
- (A) Faster kinetics
 - (B) Slower kinetics
 - (C) Same kinetics
 - (D) No transformation possible
- 68.** Alloying with Mo in steel retards pearlite formation by:
- (A) Lowering diffusion rates
 - (B) Raising hardenability
 - (C) Increasing Ms temperature
 - (D) Promoting bainite
- 69.** Under continuous cooling (CCT) versus isothermal (TTT), the effective “nose” moves to:
- (A) Longer times
 - (B) Shorter times
 - (C) Same time
 - (D) It disappears
- 70.** What is the critical cooling rate to bypass pearlite formation estimated from the TTT by $T_{austenite} = 800^{\circ}\text{C}$, $T_{nose} = 550^{\circ}\text{C}$ after 10 seconds?
- (A) 25°C/s
 - (B) 50°C/s
 - (C) 80°C/s
 - (D) 100°C/s
- 71.** Bainite forms at temperatures below the pearlite nose but above Ms. Its morphology is controlled by:
- (A) Shear plus diffusion
 - (B) Pure shear
 - (C) Pure diffusion
 - (D) Eutectic reaction
- 72.** The incubation time for austenite \rightarrow pearlite at a given temperature is inversely proportional to:
- (A) Nucleation rate
 - (B) Diffusion distance

- (C) Grain size
 - (D) Ms temperature
- 73.** In a TTT diagram, the start and finish curves for pearlite are separated because of:
- (A) Finite growth rate
 - (B) Heterogeneous nucleation only
 - (C) Spinodal decomposition
 - (D) Martensitic transformation
- 74.** A TTT curve for isothermal bainite shows two noses: upper bainite (UB) and lower bainite (LB). Which forms at lower temperature?
- (A) UB
 - (B) LB
 - (C) They coincide
 - (D) Neither forms bainite
- 75.** Jominy end-quench hardenability curves are analogous to CCT diagrams because they map:
- (A) Cooling rate versus distance
 - (B) Composition versus hardness
 - (C) Time versus temperature
 - (D) Grain size versus hardness
- 76.** Factors that shift the TTT nose to longer times include lower austenite grain size and:
- (A) Higher C content
 - (B) Lower C content
 - (C) Higher Si content
 - (D) Lower Mn content
- 77.** At temperatures just above Ms, the transformation to martensite upon quench is:
- (A) Athermal (athermite)
 - (B) Isothermal
 - (C) Diffusion-controlled
 - (D) Nucleation-controlled
- 78.** Continuous cooling transformation (CCT) diagrams differ from TTT by incorporating:
- (A) Non-isothermal paths
 - (B) Higher accuracy only

- (C) Alloy composition change
- (D) Pressure effects

79. Total porosity ϕ_t and open porosity ϕ_o differ by:

$$\phi_o = \phi_t - \phi_{\text{closed}}.$$

Closed porosity does not contribute to:

- (A) Fluid permeability
- (B) Bulk density reduction
- (C) Thermal insulation
- (D) Acoustic damping

80. The Archimedes method for porosity uses immersion to determine:

- (A) Open porosity only
- (B) Closed porosity only
- (C) Total porosity
- (D) Pore size distribution

81. Mercury intrusion porosimetry measures pore throat sizes by relating pressure P to pore diameter d via Washburn:

$$P = -\frac{4\gamma \cos \theta}{d}.$$

Higher intrusion pressure corresponds to:

- (A) Larger throats
- (B) Smaller throats
- (C) No pores
- (D) Closed pores only

82. BET adsorption can estimate surface area and microporosity by fitting the isotherm to:

$$\frac{1}{V(P_0/P - 1)} = \frac{C - 1}{V_m C} \frac{P}{P_0} + \frac{1}{V_m C}.$$

Here V_m yields:

- (A) Monolayer volume
- (B) Pore volume
- (C) Total volume
- (D) Closed porosity

83. In sintered ceramics, closed porosity reduces:

- (A) Thermal conductivity
- (B) Open-cell permeability
- (C) Surface area

(D) Weight only

84. Permeability contrast between open and closed porosity is due to:

- (A) Connectivity
- (B) Pore size only
- (C) Pore shape only
- (D) Solid stiffness

85. Effective medium models (Maxwell–Garnett) for porosity predict composite elastic modulus reduction proportional to:

$$E_{\text{eff}} = E_0(1 - a\phi).$$

Open porosity > 10% yields $a \approx$

- (A) 2–4
- (B) 0.1
- (C) 10
- (D) 0.01

86. Differential pressure porosimetry uses increasing pressure steps to resolve:

- (A) Pore size distribution
- (B) Total porosity only
- (C) Closed porosity only
- (D) Density only

87. Image analysis of micrographs for porosity quantifies area fraction which equals volume fraction in:

- (A) Isotropic microstructures
- (B) Anisotropic microstructures
- (C) Grained materials only
- (D) Liquids only

88. Pore throat to body ratio influences:

- (A) Relative permeability
- (B) True density
- (C) Closed porosity fraction
- (D) BET surface area

89. Compaction of powders reduces porosity but may trap gas, leading to:

- (A) Closed porosity
- (B) Open porosity
- (C) Zero porosity
- (D) Increased permeability

- 90.** Porosity anisotropy in extruded metals arises from:
- (A) Directional pore alignment
 - (B) Chemical composition
 - (C) Grain boundary density
 - (D) Surface finish
- 91.** The relationship between porosity and thermal conductivity in ceramics is:
- (A) Inverse nearly linear
 - (B) Direct linear
 - (C) No relation
 - (D) Exponential
- 92.** High-throughput porosity measurement by CT scanning relies on:
- (A) X-ray attenuation contrast
 - (B) Mercury intrusion
 - (C) BET adsorption
 - (D) Archimedes method
- 93.** Pore connectivity factor influences diffusion and is quantified by:
- (A) Tortuosity
 - (B) Porosity only
 - (C) Contact angle
 - (D) Grain size
- 94.** In pharmaceuticals, tablet disintegration time correlates with:
- (A) Open porosity
 - (B) Closed porosity
 - (C) Grain size
 - (D) Particle shape
- 95.** Freeze-drying introduces macroporosity by:
- (A) Sublimation of frozen solvent
 - (B) Evaporation of liquid only
 - (C) Chemical etching
 - (D) Sintering
- 96.** Porosity in polymer foams is typically achieved by:
- (A) Gas blowing agents
 - (B) Mercury intrusion
 - (C) BET adsorption
 - (D) Archimedes immersion

97. In fuel cell electrodes, hierarchical porosity enhances:
- (A) Mass transport
 - (B) Electrical conductivity only
 - (C) Thermal insulation
 - (D) Magnetic properties
98. Capillary pressure in porous media is inversely proportional to:
- (A) Pore radius
 - (B) Tortuosity
 - (C) Solid density
 - (D) Viscosity
99. Porosity change during sintering follows:
- (A) Exponential decay with time
 - (B) Linear decrease
 - (C) No change
 - (D) Oscillation
100. For a cubic crystal with $a = 4.00 \text{ \AA}$ and Cu $K\alpha$ radiation $\lambda = 1.54 \text{ \AA}$, the (200) reflection appears at a Bragg angle $\theta \approx$
- (A) 11°
 - (B) 22°
 - (C) 44°
 - (D) 68°
101. In the same cubic crystal ($a = 3.60 \text{ \AA}$), the (111) peak is at $2\theta \approx 44^\circ$ with Cu $K\alpha$. Using Mo $K\alpha$ ($\lambda = 0.71 \text{ \AA}$), the corresponding 2θ is nearest
- (A) 21°
 - (B) 31°
 - (C) 44°
 - (D) 60°
102. A tetragonal cell has $a = b = 4.00 \text{ \AA}$, $c = 5.00 \text{ \AA}$. The ratio d_{002}/d_{200} is
- (A) 0.80
 - (B) 1.00
 - (C) 1.25
 - (D) 1.50
103. A crystallographic plane intercepts the axes at $(\frac{1}{2}a, \infty, c)$. Its Miller indices (hkl) are
- (A) (102)

- (B) (2 0 1)
 - (C) (0 2 1)
 - (D) (1 2 0)
- 104.** In a BCC lattice, which family is systematically absent?
- (A) {100}
 - (B) {110}
 - (C) {200}
 - (D) {211}
- 105.** In an FCC lattice, which reflection is allowed?
- (A) (111)
 - (B) (210)
 - (C) (100)
 - (D) (310)
- 106.** In reciprocal-space units, the radius of the Ewald sphere is
- (A) $1/\lambda$
 - (B) $2\pi/\lambda$
 - (C) $\lambda/2$
 - (D) λ
- 107.** In the CsCl structure, if $f_{\text{Cs}} = f_{\text{Cl}}$ exactly, which (hkl) would be extinct?
- (A) (110)
 - (B) (200)
 - (C) (111)
 - (D) (002)
- 108.** The strongest non-origin peak in a Patterson map corresponds to
- (A) the self-vector at (0,0,0)
 - (B) the shortest interatomic vector
 - (C) the largest interatomic distance
 - (D) the average atomic position
- 109.** In a Williamson–Hall plot of $\beta \cos \theta$ vs. $\sin \theta$, the intercept on the vertical axis yields information on
- (A) microstrain
 - (B) crystallite size
 - (C) instrumental broadening
 - (D) lattice parameter
- 110.** In space group $\text{Pm}\bar{3}\text{m}$ (primitive cubic), which reflection is systematically absent?

- (A) (100)
 - (B) (110)
 - (C) (111)
 - (D) none
111. Which plane family has the highest multiplicity in a cubic lattice?
- (A) {100}
 - (B) {110}
 - (C) {111}
 - (D) {210}
112. At fixed θ , heavier elements scatter X-rays more strongly because their atomic scattering factor f
- (A) drops off faster with angle
 - (B) has a larger magnitude
 - (C) is constant
 - (D) is purely imaginary
113. Hydrogen has a negative coherent neutron scattering length, which causes
- (A) constructive interference
 - (B) destructive interference
 - (C) no contrast
 - (D) resonance scattering
114. The “phase problem” in crystallography refers to the inability to measure directly the
- (A) structure-factor amplitudes
 - (B) structure-factor phases
 - (C) unit-cell dimensions
 - (D) Bragg angles
115. A key advantage of powder diffraction over single-crystal methods is that
- (A) it requires no single crystal
 - (B) it gives directional intensities
 - (C) it solves phases directly
 - (D) it has no peak overlap
116. In a Debye–Scherrer camera of radius L , the ring radius R for a reflection at 2θ satisfies
- (A) $R = L \sin 2\theta$
 - (B) $R = L \sin \theta$
 - (C) $R = L \tan 2\theta$

(D) $R = 2L \tan \theta$

- 117.** A rocking-curve (ω -scan) width in single-crystal XRD primarily measures the crystal's

- (A) lattice parameter
- (B) mosaic spread
- (C) thermal vibration amplitude
- (D) unit-cell volume

- 118.** The reciprocal-lattice metric for a hexagonal cell gives

$$\frac{1}{d^2} = \frac{4}{3} \frac{h^2 + hk + k^2}{a^2} + \frac{l^2}{c^2}.$$

For reflections with $l = 0$, which term dominates when $a \ll c$?

- (A) the first (basal-plane) term
- (B) the second (l -dependent) term
- (C) both equally
- (D) neither

- 119.** For a hexagonal crystal with $a = 3.00 \text{ \AA}$, the d -spacing of $(10\bar{1}0)$ is closest to

- (A) 2.30 \AA
- (B) 2.60 \AA
- (C) 3.00 \AA
- (D) 3.50 \AA

- 120.** A real-space cell has volume $V = 100 \text{ \AA}^3$. The reciprocal-cell volume V^* is

- (A) 0.001 \AA^{-3}
- (B) 0.01 \AA^{-3}
- (C) 0.1 \AA^{-3}
- (D) 1 \AA^{-3}

- 121.** If $2W = 0.10$ in Debye–Waller theory, the diffracted intensity is reduced to approximately

- (A) 90%
- (B) 50%
- (C) 10%
- (D) 1%

- 122.** A powder pattern with d -spacing ratios $1 : \frac{1}{\sqrt{2}} : \frac{1}{\sqrt{3}} : \dots$ corresponds to which lattice?

- (A) Simple cubic
- (B) BCC

- (C) FCC
 - (D) Tetragonal
- 123.** In the diamond structure (space group $Fd\bar{3}m$), which reflection is forbidden?
- (A) (111)
 - (B) (220)
 - (C) (200)
 - (D) (222)
- 124.** In Laue diffraction, the condition $\mathbf{k} - \mathbf{k}_0 = \mathbf{G}$ implies that
- (A) $|\mathbf{k}| = |\mathbf{k}_0|$
 - (B) $\mathbf{G} \perp \mathbf{k}_0$
 - (C) \mathbf{k} is parallel to \mathbf{k}_0
 - (D) $\mathbf{G} = 0$
- 125.** The Scherrer constant K depends on crystallite shape; for spherical crystals $K \approx$
- (A) 0.5
 - (B) 0.9
 - (C) 1.5
 - (D) 2.0
- 126.** In neutron diffraction, contrast between isotopes arises because
- (A) different nuclear scattering lengths
 - (B) different atomic numbers
 - (C) different electron densities
 - (D) different magnetic moments only
- 127.** Electron diffraction at 200 kV ($\lambda \approx 0.025 \text{ \AA}$) yields nearly planar Ewald sphere, so zone-axis patterns show
- (A) only one reflection
 - (B) many simultaneous reflections
 - (C) no reflections
 - (D) purely inelastic scattering
- 128.** Kikuchi lines in TEM diffraction arise primarily from
- (A) elastic Bragg scattering
 - (B) inelastic diffraction of thermal diffuse scattering
 - (C) fluorescent X-rays
 - (D) electron-electron scattering
- 129.** The dynamical theory of electron diffraction predicts that

- (A) forbidden reflections never appear
 - (B) intensities deviate from $|F|^2$
 - (C) λ depends on thickness
 - (D) phases are directly measured
- 130.** Convergent-beam electron diffraction (CBED) patterns contain higher-order Laue zone (HOLZ) rings which can be used to determine
- (A) lattice strain only
 - (B) absolute specimen thickness and symmetry
 - (C) only chemical composition
 - (D) only unit-cell volume
- 131.** A crystal's mosaic spread will broaden Bragg peaks in
- (A) powder Debye-Scherrer
 - (B) single-crystal rocking curves
 - (C) Laue back-reflection
 - (D) all of the above
- 132.** The Debye scattering equation for total scattering (including diffuse) involves a sum over
- (A) only Bragg peaks
 - (B) all interatomic pairs
 - (C) only nearest neighbors
 - (D) only unit-cell edges
- 133.** The R factor in crystallographic refinement is defined as
- (A) $\sum |F_{\text{obs}} - F_{\text{calc}}| / \sum F_{\text{obs}}$
 - (B) $\sum |I_{\text{obs}} - I_{\text{calc}}| / \sum I_{\text{obs}}$
 - (C) $\sum (F_{\text{obs}}^2 - F_{\text{calc}}^2) / \sum F_{\text{obs}}^2$
 - (D) all of the above (different conventions)
- 134.** Systematic absences for a C-centred orthorhombic cell require $h + k =$
- (A) even
 - (B) odd
 - (C) multiple of 3
 - (D) any integer
- 135.** In electron diffraction, the extinction distance is the thickness over which
- (A) a diffracted beam is completely extinguished by absorption
 - (B) primary and diffracted beams exchange intensity
 - (C) all reflections overlap

- (D) only inelastic scattering occurs
- 136.** A crystallographic twin law relates two domains by
- (A) a rotation or reflection symmetry operation not in the point group
 - (B) a simple translation
 - (C) an arbitrary strain
 - (D) identical orientation
- 137.** In powder indexing, the successively observed d -spacings of NaCl follow ratios $1 : \frac{1}{\sqrt{2}} : \frac{1}{\sqrt{3}} : \frac{1}{2} : \dots$. The fifth observed peak corresponds to which (hkl) ?
- (A) (220)
 - (B) (200)
 - (C) (210)
 - (D) (211)
- 138.** The ratio of integrated intensities I_{111}/I_{200} for an A1-type fcc crystal (one atom per lattice point) neglecting temperature and Lorentz factors is
- (A) 1:1
 - (B) 2:1
 - (C) 3:2
 - (D) depends only on atomic form factor ratio $f(111)/f(200)$
- 139.** In a reflection-geometry single-crystal experiment, the ω angle is the rotation about an axis
- (A) perpendicular to the diffracting plane
 - (B) parallel to the incident beam
 - (C) through the detector
 - (D) none of the above
- 140.** Anisotropic displacement parameters (ADPs) refine atomic vibrations as
- (A) spherical shells
 - (B) ellipsoids
 - (C) rigid rotations
 - (D) constant offsets
- 141.** The Patterson superposition method uses known heavy-atom positions to estimate
- (A) light-atom positions
 - (B) unit-cell dimensions
 - (C) symmetry operators
 - (D) absorption coefficients

- 142.** In a primitive orthorhombic cell, the number of symmetry-equivalent positions under point group mmm is
- (A) 1
 - (B) 2
 - (C) 4
 - (D) 8
- 143.** The Guinier approximation in small-angle X-ray scattering is valid for $qR_g < 1$, where $q = 4\pi \sin \theta / \lambda$ and R_g is the radius of gyration. It predicts $\ln I(q) \propto -q^2 R_g^2 / 3$. This allows direct extraction of
- (A) crystal size
 - (B) particle shape
 - (C) radius of gyration
 - (D) lattice parameter
- 144.** Relative density (specific gravity) compares the density of a material to that of water. A metal with relative density 7.8 has a density of approximately:
- (A) 0.78 g/cm³
 - (B) 7.8 g/cm³
 - (C) 78 g/cm³
 - (D) 0.078 g/cm³
- 145.** On the Mohs hardness scale, which mineral is harder than quartz (hardness 7)?
- (A) Orthoclase feldspar
 - (B) Apatite
 - (C) Topaz
 - (D) Talc
- 146.** The Rockwell hardness test measures indentation depth under a fixed load. A higher Rockwell number indicates:
- (A) Greater indentation depth
 - (B) Smaller indentation depth
 - (C) Larger indenter diameter
 - (D) Lower applied load
- 147.** Young's modulus is defined as the ratio of:
- (A) Stress to strain in the elastic region
 - (B) Strain to stress in the plastic region
 - (C) Load to cross-sectional area
 - (D) Deflection to applied moment

- 148.** Flexural strength (modulus of rupture) is determined from a three-point bend test by:
- (A) Maximum tensile stress at fracture
 - (B) Maximum compressive stress at fracture
 - (C) Maximum shear stress at mid-span
 - (D) Average stress across the section
- 149.** In a three-point bend specimen of rectangular cross-section, the flexural strength increases if:
- (A) Span length increases
 - (B) Specimen thickness decreases
 - (C) Applied load at fracture increases
 - (D) Width decreases
- 150.** Compressive strength is defined as the maximum compressive stress sustained before:
- (A) Elastic limit
 - (B) Yielding
 - (C) Failure
 - (D) Necking
- 151.** Fracture toughness K_{IC} characterizes a material's resistance to:
- (A) Stable crack growth under mode I loading
 - (B) Plastic deformation at dislocations
 - (C) Surface hardness
 - (D) Creep at high temperature
- 152.** A material with low fracture toughness is likely to exhibit:
- (A) Ductile fracture
 - (B) Brittle fracture
 - (C) Superplasticity
 - (D) High fatigue life
- 153.** Brittle fracture often occurs with little prior plastic deformation because:
- (A) Crack tip blunting is extensive
 - (B) Energy is absorbed in dislocation motion
 - (C) Crack propagates rapidly without yielding
 - (D) Voids nucleate and coalesce first
- 154.** Engineering stress is defined as load divided by the original cross-sectional area. True strain is defined as:
- (A) $\Delta L/L_0$

- (B) $\ln(L/L_0)$
- (C) $(L - L_0)/L$
- (D) $L/L_0 - 1$

155. In a tensile test, the linear elastic region ends at the:

- (A) Ultimate tensile strength
- (B) Yield strength
- (C) Fracture point
- (D) Proportional limit

156. For steel, Poisson's ratio is typically about:

- (A) 0.1
- (B) 0.3
- (C) 0.5
- (D) 1.0

157. The yield strength of a metal is defined at a specified offset, commonly:

- (A) 0.01%
- (B) 0.1%
- (C) 1%
- (D) 10%

158. Hardness correlates empirically with tensile strength because:

- (A) Both measure resistance to elastic deflection
- (B) Both involve plastic flow under load
- (C) Both are independent of microstructure
- (D) Both are measured on Vickers scale

159. A material shows 200 HV (Vickers hardness). Its approximate tensile strength is:

- (A) 200 MPa
- (B) 600 MPa
- (C) 1000 MPa
- (D) 2000 MPa

160. Creep is time-dependent deformation under constant stress. Primary creep exhibits:

- (A) Increasing strain rate
- (B) Decreasing strain rate
- (C) Constant strain rate
- (D) Rapid failure

- 161.** Fatigue limit of steel refers to the stress amplitude below which:
- (A) Plastic deformation accumulates
 - (B) No fatigue failure occurs
 - (C) Creep dominates
 - (D) Brittle fracture initiates
- 162.** An Einstein solid with an Einstein temperature Θ_E has $C_V/R = 0.5$ at $T = \Theta_E$. What is the approximate value of C_V/R at $T = 2\Theta_E$?
- (A) 0.6
 - (B) 1.0
 - (C) 2.0
 - (D) 2.5
- 163.** Silver iodide undergoes an “order-disorder” transition with a sharp heat-capacity anomaly near its superionic transition at 420 K. This peak in C_p arises primarily from:
- (A) Lattice vibrations only
 - (B) Electronic excitations
 - (C) Configurational entropy of Ag^+ mobility
 - (D) Magnetic ordering
- 164.** In an ionic solid at room temperature, the dominant contribution to thermal conductivity is from phonons. Which mechanism most limits κ at high T ($T \gtrsim \Theta_D/2$)?
- (A) Phonon-electron scattering
 - (B) Phonon-phonon Normal processes
 - (C) Phonon-phonon Umklapp processes
 - (D) Boundary scattering
- 165.** Silicon has $\alpha = 2.6 \times 10^{-6} \text{ K}^{-1}$ at 300 K, while germanium has $\alpha = 6 \times 10^{-6} \text{ K}^{-1}$. This difference largely reflects:
- (A) Mass difference only
 - (B) Bond anharmonicity differences
 - (C) Electronic contributions
 - (D) Gravitational effects
- 166.** Garnet ($\text{Y}_3\text{Al}_5\text{O}_{12}$) has a low thermal expansion $\alpha \approx 5 \times 10^{-6} \text{ K}^{-1}$. Such low α in a complex oxide is due to:
- (A) Strong covalent Al-O bonds
 - (B) Weak ionic Y-O bonds
 - (C) Magnetic ordering
 - (D) Porosity

- 167.** In a metal at cryogenic temperature ($T \ll \Theta_D$), the electronic thermal conductivity κ_e dominates. According to the Wiedemann–Franz law $\kappa_e/\sigma T = L$, where L is the Lorenz number. A deviation from constant L indicates:
- (A) Phonon drag on electrons
 - (B) Pure electron scattering
 - (C) Magnetic impurities only
 - (D) Perfect lattice
- 168.** Diamond has $k \sim 2000 \text{ W m}^{-1} \text{ K}^{-1}$, $\rho = 3500 \text{ kg/m}^3$, $c_p = 500 \text{ J kg}^{-1} \text{ K}^{-1}$. Its thermal diffusivity is roughly:
- (A) $10^{-8} \text{ m}^2/\text{s}$
 - (B) $10^{-6} \text{ m}^2/\text{s}$
 - (C) $10^{-4} \text{ m}^2/\text{s}$
 - (D) $10^{-2} \text{ m}^2/\text{s}$
- 169.** A polymer exhibits a jump in heat capacity at its glass transition T_g . This ΔC_p reflects increased segmental mobility. For a 100 g mol^{-1} polymer with $\Delta C_p = 0.5 \text{ J g}^{-1} \text{ K}^{-1}$, the molar jump is:
- (A) $5 \text{ J mol}^{-1} \text{ K}^{-1}$
 - (B) $50 \text{ J mol}^{-1} \text{ K}^{-1}$
 - (C) $500 \text{ J mol}^{-1} \text{ K}^{-1}$
 - (D) $5000 \text{ J mol}^{-1} \text{ K}^{-1}$
- 170.** In tungsten carbide (WC), the thermal expansion coefficient is $\alpha = 5 \times 10^{-6} \text{ K}^{-1}$, whereas for cobalt binder $\alpha = 13 \times 10^{-6} \text{ K}^{-1}$. Under rapid heating, this mismatch causes:
- (A) Compressive stress in WC
 - (B) Tensile stress in WC
 - (C) No stress
 - (D) Shear failure only
- 171.** Zirconium tungstate (ZrW_2O_8) exhibits isotropic negative thermal expansion over a broad temperature range. This counter-intuitive behavior is primarily due to:
- (A) Rigid-unit transverse vibrations of WO_4 and ZrO_6 polyhedra
 - (B) Electronic delocalization upon heating
 - (C) Cation–anion size-mismatch strain relief
 - (D) Magnetostrictive domain reorientation
- 172.** In diamond, the presence of the ^{13}C isotope (1.1 % natural abundance) introduces phonon scattering that limits its thermal conductivity. The scattering rate due to isotopic disorder scales with the mass variance $(\Delta M/M)^2$. This type of phonon scattering is known as:

- (A) Mass-defect (isotope) scattering
- (B) Phonon–electron scattering
- (C) Phonon–phonon Umklapp scattering
- (D) Grain-boundary (surface) scattering