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EFFECT OF STRAIN RATE ON

THE TENSILE PROPERTIES OF

IRRADIATED INCONEL 718

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J. M. Steichen

A. L. Ward

January, 1976

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J. M. Steichen and A. L. Ward

ABSTRACT

The effect of strain rate on the tensile properties of neutron irradiated Inconel 718 has been experimentally determined. Specimens were irradiated in the Experimental Breeder Reactor-II to total neutron fluences of 0.6 to 1.0 x 10^{22} n/cm² (at 400° C), 1.0 to 2.4 x 10^{22} n/cm² (at 593° C) and 3.1 to 3.4 x 10^{22} n/cm² (at 649° C). Test conditions included nominal strain rates from 3×10^{-5} to 1sec 1 and temperatures from 232 to 649°C. For the temperature range of this study the strength and ductility of unirradiated material were essentially insensitive to increasing strain rate. Neutron irradiation at a temperature of 400°C did not alter this behavior since both strength and ductility remained essentially unaffected by increasing strain rate. At the higher irradiation temperatures (593°C and above) strength properties were consistent with unirradiated behavior and pre- and postirradiation ductility values were essentially identical at the lower test temperatures (< 550°C). At test temperatures above 550°C all irradiated specimens exhibited very low ductility as a result of helium embrittlement. Increasing strain rate at these temperatures reduced the effect of this embrittlement and resulted in ductility values approaching those observed prior to irradiation.

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INTRODUCTION

Inconel 718, a precipitation hardenable nickel-base alloy, is being used in a variety of structural components in liquid metal fast breeder reactor systems. Applications in the various components include thermal liners, springs, connectors, shear pins, and wear pads. Service conditions potentially include temperatures to 650° C and total neutron fluences to greater than 1 x 10^{23} n/cm². Mechanical properties data representative of these service conditions are needed for safety analyses and operations guidance for the associated structural components. Of special significance are irradiation effects data representative of postulated accident conditions.

The elevated temperature mechanical properties of unirradiated Inconel 718 have been extensively characterized (1,3) and limited data are available on the effects of neutron irradiation on these properties (4,6). Claudson (4) has shown that irradiation (to 1 x 10^{21} n/cm², E > 1 MeV) in a thermal reactor significantly reduces elevated temperature (740°C) tensile ductility with little effect on strength and Bohm et al. (5) demonstrated that thermal-reactor irradiation at a fluence of 8.1 x 10^{21} n/cm² (E > 1 MeV) significantly reduces the 650°C rupture life and ductility. Recently, Ward et al. (6) reported tensile data for the subject material which had been irradiated in a fast reactor (EBR-II) at temperatures to 649°C and total fluences to 3.4 x 10^{22} n/cm². These results revealed that irradiation did not greatly alter strength properties; however, ductility was severely reduced when test temperatures approached 649°C. Of the available information, none indicates the effect of strain rate on the tensile properties of irradiated material.

In the present study, the effect of strain rate on the tensile properties of unirradiated and irradiated Inconel 718 has been determined. Irradiation conditions included fast fluences of 1.0 x 10^{22} n/cm² (at 391°C), 2.4 x 10^{22} n/cm² (at 593°C), and 3.4 x 10^{22} n/cm² (at 649°C). Test parameters include strain rates to 1.0 sec⁻¹ and temperatures to 649°C.

EXPERIMENTAL PROCEDURE

Test Material and Specimens

The Inconel 718 used in this study was obtained in the form of 1.27 cm thick plate. Miniature buttonhead tensile specimens (Figure 1) were fabricated from the longitudinal forming direction of the plate and subsequently used for control testing and the irradiation experiment. Heat treatment of the test material consisted of annealing at 954°C for 0.5 hours followed by aging at 717°C for 8 hours, furnace cooling to 621°C, and holding at 621°C until the total aging time reached 18 hours. The chemical composition for the test material is presented in Table I.

TABLE I

CHEMICAL COMPOSITION OF INCONEL 718-HEAT 52C9EK (Wt. %	CHEMICAL	COMPOSITION	0F	INCONEL	718-HEAT	52C9EK	(Wt.	%
--	----------	-------------	----	---------	----------	--------	------	---

C - 0.07	A1 - 0.59
Mn - 0.08	Ti - 1.08
Fe - 18.19	P - 0.007
S - 0.007	Co - 0.03
Si - 0.17	Mo - 0.03
Cu - 0.11	B - 0.0032
Ni - 53.44	Cb + Ta - 5.10
Cr - 18.11	

Irradiation Experiment

The Experimental Breeder Reactor-II (EBR-II) at the Idaho National Engineering Laboratory was used for the irradiation experiment. The experiment was conducted in a Row 2 core position in structural materials subassemblies designated X-195 and X-198. All test specimens were irradiated in a sodium environment. Irradiation conditions included the following total fluences and temperatures: 0.5 to 1.0 x 10^{22} n/cm² at 391 to 404°C, 1.0 to 2.4 x 10^{22} n/cm² at 593°C, and 3.1 to 3.4 x 10^{22} n/cm² at 649°C.

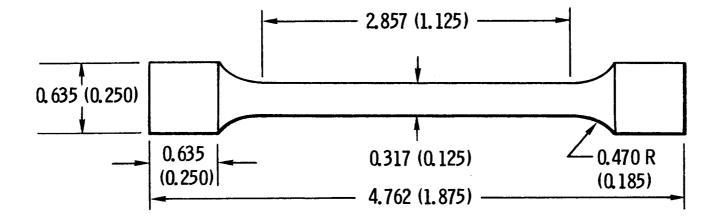


FIGURE 1. Tensile Specimen Geometry. Dimensions in cm (in.).

Test Procedure

All tests were performed on hard beam machines. An Instron Tensile Machine was utilized for all tests at strain rates to $7.4 \times 10^{-3} \text{ sec}^{-1}$ and an MTS Electro-Hydraulic test machine for strain rates from 7.4×10^{-3} to 1.0 sec^{-1} . Both machines were operated at constant crosshead velocity and strain rate values given here are based on the crosshead velocity and initial specimen gauge length. Ductility data reported in this report were obtained from pre- and post-test specimen measurements.

RESULTS AND DISCUSSION

Unirradiated Behavior

Tensile testing of unirradiated Inconel 718 was performed at temperatures from room temperature to 760°C and strain rates from 3 x 10^{-5} to 1.0 sec⁻¹. Results of these tests provided a description of unirradiated tensile properties and also established a basis for ascertaining the effects of neutron irradiation. Tensile property data obtained from each test included the 0.2% yield and ultimate strengths, total and uniform elongations, and reduction of area. These data are tabulated in Table II.

The temperature and strain rate dependence of the ultimate strength of unirradiated Inconel 718 is presented in Figure 2. At any given rate the ultimate strength decreases with increasing temperature. At room temperature a positive strain rate sensitivity is observed, i.e., the ultimate strength increases with increasing strain rate. Increasing test temperature to 232°C alters this behavior sinch a slightly negative strain rate sensitivity is observed. At temperatures of 427 and 538°C the ultimate strength is essentially insensitive to strain rate. For temperatures of 649 and 760°C there is a clear departure from the strain rate dependence observed at the lower temperatures. For strain rates below about $10^{-3}\ \text{sec}^{-1}$ (at 649° C) and 10^{-2} sec⁻¹ (760°C) the ultimate strength is strongly strain rate dependent. Such dependence, reported previously for other austenitic materials (7,8) has been attributed to the dominance of creep deformation at these temperatures and low strain rates. As strain rate is increased, creep (time and temperature dependent) is progressively restricted until it no longer exerts a pronounced influence at the higher strain rates.

TABLE II

ELEVATED TEMPERATURE TENSILE PROPERTIES OF INCONEL 718 AT VARIOUS STRAIN RATES

Temp. °C	° -1 Sec -1	St PEL	rength - MPa 0,2% Yie		ltimate	Elonga Total	tion-% Uniform	Reduction of Area - %	Site Sequence
22	3 x 10 ⁻⁵	785 (113.8) 840 (121.9) 816 (118.4) 814 (118.0)	994 (144 1053 (15	4.5) 1365 4.2) 1267 2.7) 1318 0.5) 1326	(198.0) (183.7) (191.1) (192.3)	16.4 20,0 19.0 19.5	15.5 18.8 18.2 18,0	30.9 30.9 28.5 38.2	HO 4781 4782 4794 4796
SUPPLEMENT TO A	3 x 10 ⁻³ 6 x 10 ⁻² 1,0	1096 (159.0) 1031 (149.6) 929 (134.8)	1177 (17)	8,0) 1393 0,7) 1413 9,3) 1404	(202.0) (204.9) (203.7)	15.6 15.0 14,5	10.9 8.3 10.0	31.5 31.7 36.1	4862 4833 4831
232	3 x 10 ⁻⁵ 3 x 10 ⁻³ 6 x 10 ⁻² 1.0	735 (106,6) 836 (125,9) 840 (121,9) 918 (133,2) 833 (120,8)	1054 (15) 1034 (15) 1078 (15)	5.6) 1314 2.8) 1284 0.0) 1278 6.4) 1255 5.0) 1218	(190.6) (186.3) (185.4) (182.0) (176.7)	14.2 16.2 15.3 13.8 16.1	12.9 11.8 12.4 9.3 11.5	22.1 29.0 35.8 32.8 37.5	4812 4803 4863 4834 4832
427	3 x 10 ⁻⁵ 3 x 10 ⁻³ 6 x 10 ⁻² 1.0	774 (112.2) 751 (108.9) 840 (121.9) 820 (118.9) 942 (136,6)	919 (13: 1006 (14: 967 (14:	5.1) 1147 3.3) 1131 5.9) 1197 0,3) 1201 4,3) 1171	(1664) (164,1) (173,6) (174.2) (169.9)	17.4 17.9 13.3 12.2 10,8	13.5 14.4 11.1 11.0 10.0	30,9 30,9 33,3 29.8 35.0	4784 4785 4864 4828 4832
538	3×10^{-5} 3×10^{-3} 6×10^{-2} 1.0	673 (97.6) 718 (104.1) 791 (114.7) 814 (118.0 838 (121.6) 800 (116.1)	889 (129 991 (149 971 (149 974 (149	0.1) 1120 9.0) 1104 3.8) 1187 0.9) 1158 1.2) 1133 5.5) 1134	(162,4) (160,1) (172,1) (168,0) (164,4) (164,5)	17,9 13.6 12.5 12,7 12.8 12,8	15.5 12.2 10.9 9.4 8.9 11.2	30.1 25.2 32.0 32.8 32.0 31.5	4786 4787 4865 4866 4829 4837
593	3×10^{-5} 3×10^{-3} 6×10^{-2} 1.0	712 (103.2) 701 (101.6) 796 (115.4) 796 (115.4) 863 (125.2) 647 (93.9)	891 (129 973 (149 964 (139 965 (149	0.1) 1108 9.3) 1101 1.1) 1171 9.8) 1166 0.0) 1122 1.4) 1129	(160.7) (159.7) (169.9) (169.1) (162.8) (163.7)	15.0 14.3 11.6 12.1 13.7 14.0	14.0 12.9 9.2 8.8 11.9 10.5	25.2 24.4 25.2 32.5 32.0 33.1	4788 4790 4867 4868 4825 4826

TABLE II

ELEVATED TEMPERATURE TENSILE PROPERTIES OF INCONEL 718 AT VARIOUS STRAIN RATES (Continued)

Temp. °C	€ - Sec-1	St PEL	trength - MPa (Ksi) 0.2% Yield Ultimate			ltimate	Elon Total i	gation-% Uniform	Reduction of Area - %	Site Sequence
649	3×10^{-5} 3×10^{-3} 6×10^{-2} 1.0	616 (89.4) 673 (97.6) 736 (106.7) 778 (112.9) 855 (124.0)	826 855 965 937 938	(119.8) (124.0 (129.9) (135.9) (136.0)	939 962 1101 1071 1092	(136.2) (139.5) (159.7) (155.3) (158.4)	6.8 7.2 14.0 15.1 11.4	6.3 6.8 9.8 11.9 8.9	17.1 15.4 28.5 30.6 30.4	HO 4792 4793 4869 4870 4830
760	1.0 3 x 10 ⁻⁵ 3 x 10 ⁻³ 6 x 10 ⁻² 1.0	826 (119.8) 471 (68.3) 493 (71.5) 628 (91.1) 706 (102.4) 620 (89.7)	926 610 621 770 787 759	(134.3) (88,5) (90.0) (111.7) (114.1) (110.1)	1096 621 632 870 967 984	(158.9) (90.1) (91.6) (126.2) (140.3) (142.7)	13.4 4.2 8.4 7.4 15.5 13.4	8.3 2.3 2.8 4.4 8.9 11.3	31.5 13.8 15.4 11.4 25.8 29.8	4827 4789 4791 4871 4836 4835

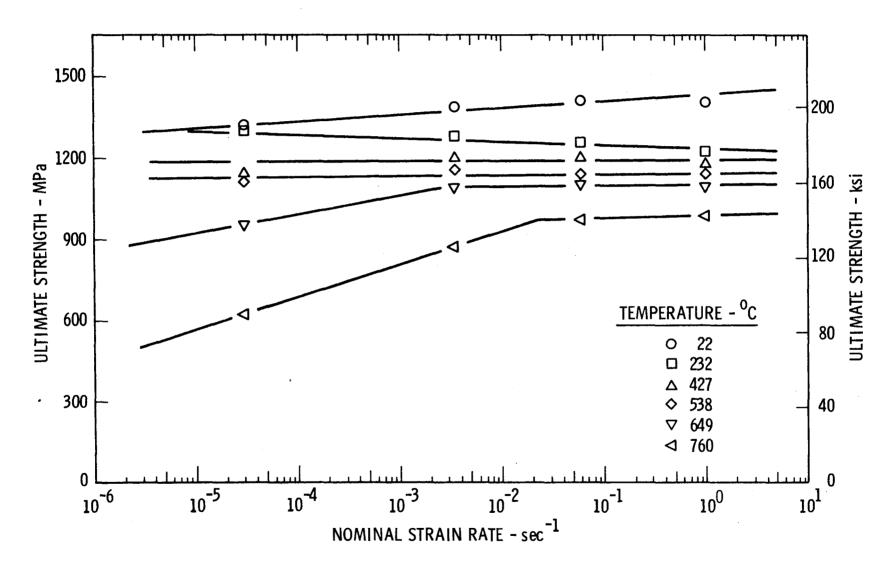


FIGURE 2. Temperature and Strain Rate Dependence of the Ultimate Strength of Unirradiated Inconel 718.

The temperature dependence of the 0.2% yield strength is similar to that of the ultimate strength in that both decrease with increasing test temperatures (Figure 3). Increasing the strain rate at temperatures between room temperature and 538°C increases the yield strength only slightly while at temperatures of 649 and 760°C it is increased substantially. Behavior at these higher temperatures is consistent with that observed for the ultimate strength.

The ductility (total and uniform elongation and reduction of area) of unirradiated material is influenced by strain rate at the various temperatures as shown in Figures 4-6. For temperatures to 538°C, both total and uniform elongation (elongation at maximum load) moderately decrease with increasing strain rate. At the higher test temperatures (649 and 760°C) both properties increase thus producing a tendency toward convergence with the low temperature results. Reduction of area increases with strain rate at all temperatures (Figure 6). As with total and uniform elongation, the largest increases in reduction of area occur at the higher temperatures.

As previously mentioned, the strain-rate response of Inconel 718 is similar to that reported for other austenitic materials. For test temperatures from 232 to 538°C the ultimate strength and ductility decrease slightly or are unaffected by increasing strain rate. This decrease has been observed in other austenitic materials and has been related to dynamic strain aging $^{(8)}$. Dynamic strain aging is characterized by serrated stress-strain curves and results from dislocation interactions with interstitial carbon and nitrogen atoms $^{(9)}$. Its influence on tensile behavior is to enhance both strength and ductility. Increasing strain rate has been shown to restrict dynamic strain aging thus producing lower strength and ductility.

In the present study extensive dynamic strain aging was observed (as evidenced by serrations) in the autographic records of tests performed at the lowest strain rate and temperatures of 232, 427, and 538°C (Figure 7). Additionally, the magnitude of the serrations was largest at 538°C (> 40 MPa) and non-existant at 649°C. Increasing strain rate altered the serrations at temperatures of 427 and 649°C as illustrated in Figures 8 and 9.

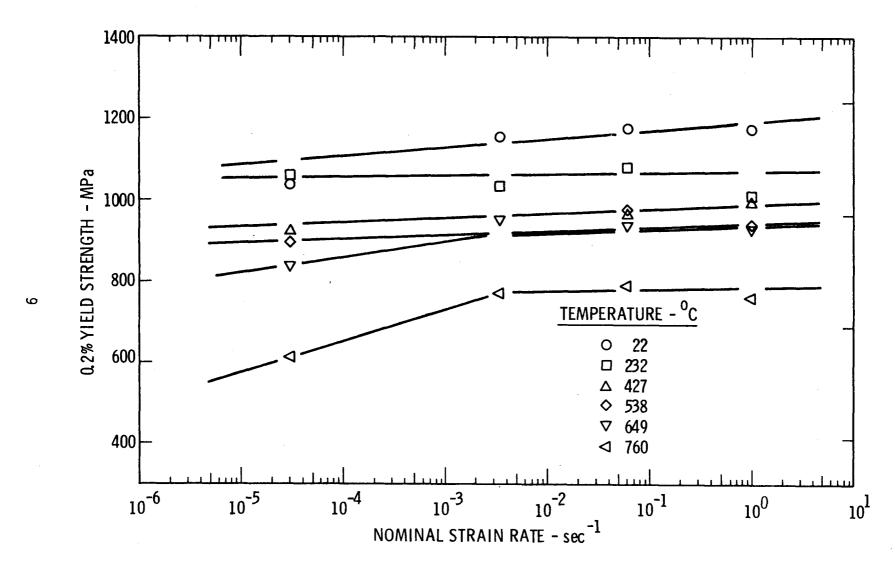


FIGURE 3. Temperature and Strain Rate Dependence of the 0.2% Yield Strength of Unirradiated Inconel 718.

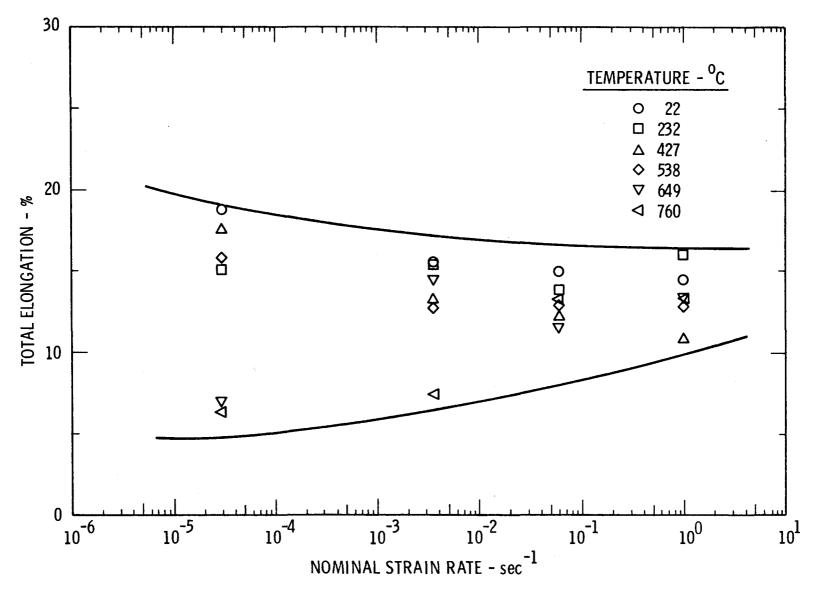


FIGURE 4. Temperature and Strain Rate Dependence of the Total Elongation of Unirradiated Inconel 718.



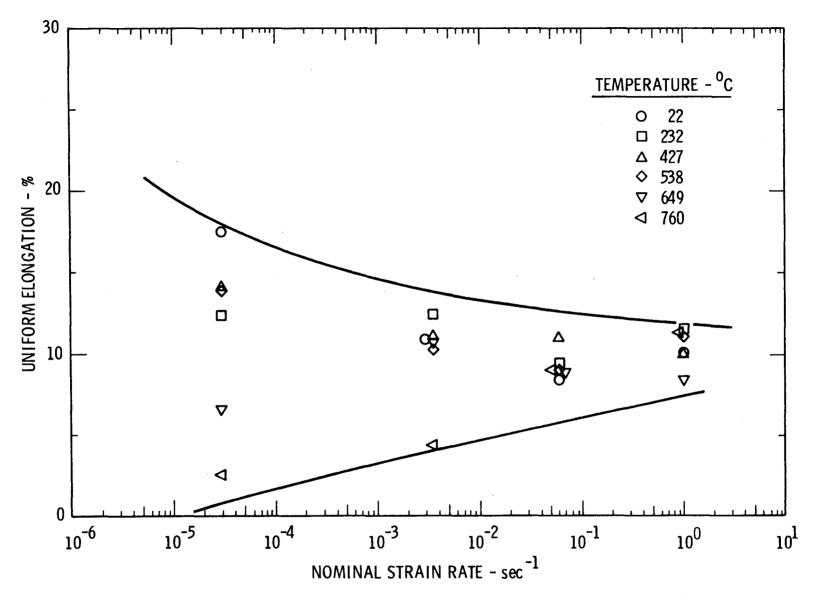


FIGURE 5. Temperature and Strain Rate Dependence of the Uniform Elongation of Unirradiated Inconel 718.

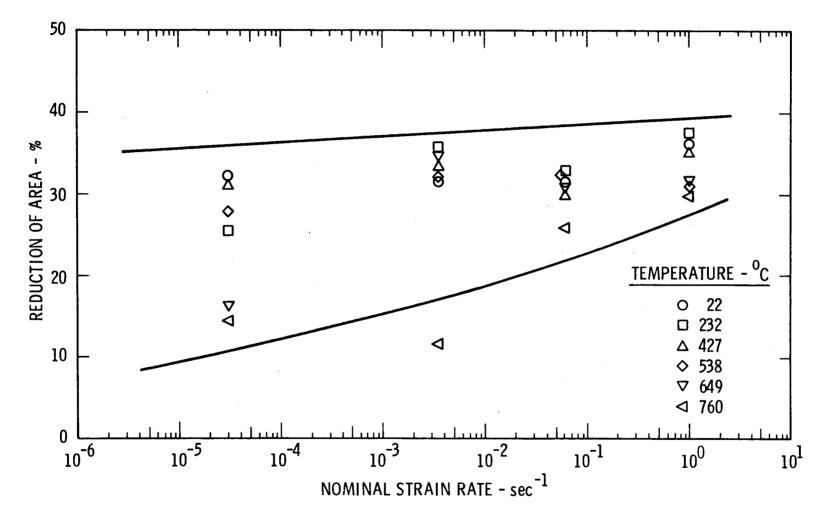


FIGURE 6. Temperature and Strain Rate Dependence of the Reduction of Area of Unirradiated Inconel 718.

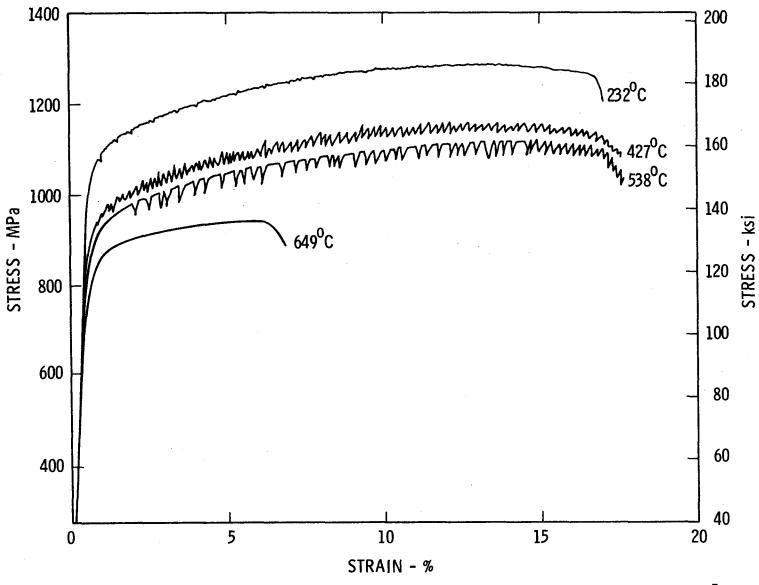


FIGURE 7. Stress-Strain Behavior of Unirradiated Inconel 718 at a Strain Rate of 3 \times 10⁻⁵ sec⁻¹.

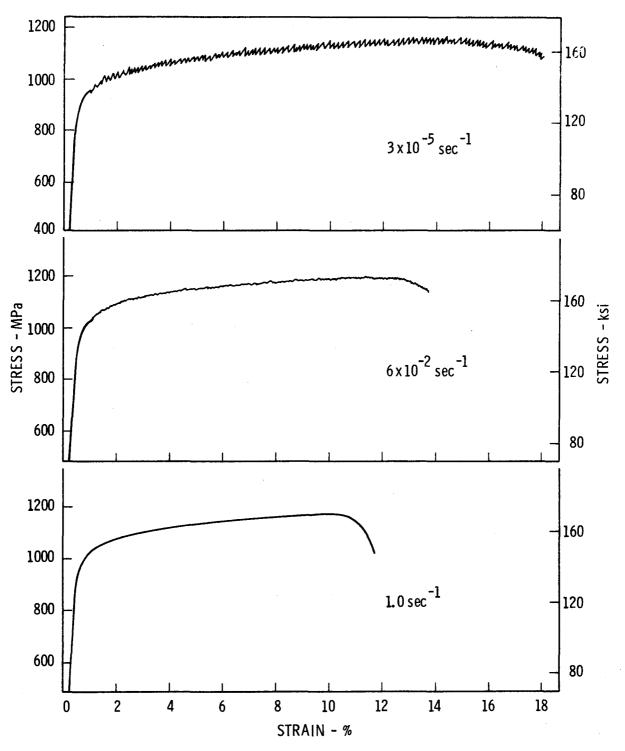


FIGURE 8. Effect of Strain Rate on the Stress-Strain Behavior of Unirradiated Inconel 718 at 427°C.

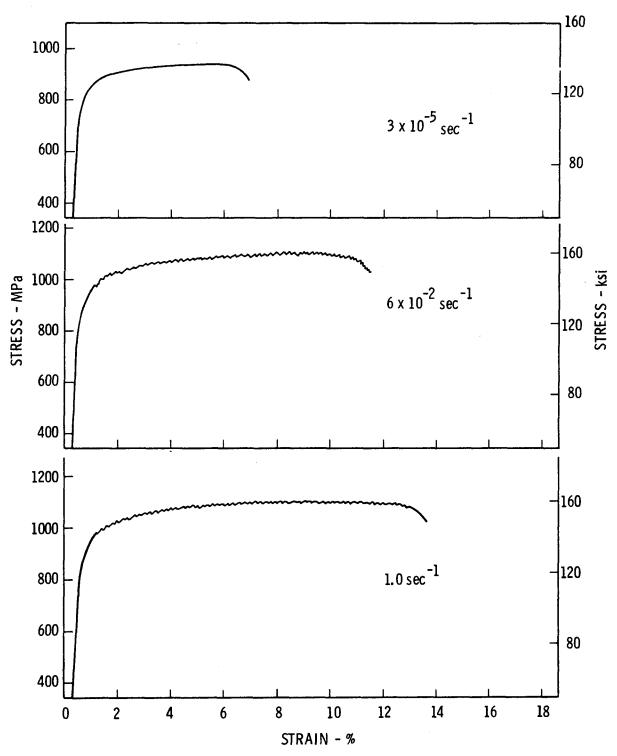


FIGURE 9. Effect of Strain Rate on the Stress-Strain Behavior of Unirradiated Inconel 718 at 649°C.

At the lowest strain rate $(3 \times 10^{-5} \text{ sec}^{-1})$ and 427°C, extensive serrations were observed in the stress-strain curve (Figure 8). Increasing strain rate to $6 \times 10^{-2} \text{ sec}^{-1}$ reduced the magnitude of the serrations, and reduced the ductility, but had little or no effect on strength properties. At a strain rate of 1.0 sec⁻¹ the serrations were essentially eliminated and the total and uniform elongation values further reduced. Therefore, it is thought that the stress-strain behavior observed at 1.0 sec⁻¹ would not be changed significantly by further increases in strain rate.

At 649°C, serrations were not observed at the lowest test rate (Figure 9). In this temperature-strain rate region, creep controls deformation and dynamic strain aging does not greatly influence properties. A minimal influence would be expected because the temperature is high enough and strain rate low enough to allow interstitial atoms to migrate along with dislocations $^{(9)}$. As strain rate increases, the interstitial atoms retard dislocation movement. This results in the flow curve serrations observed for a rate of 6 x 10^{-2} sec⁻¹ in Figure 9. Increasing the strain rate to 1.0 sec⁻¹ produced slight serrations and total and uniform elongation values slightly greater than those observed at 6 x 10^{-2} sec⁻¹.

Irradiated Behavior

Test parameters for the irradiated material included temperatures from 232 to 649°C and strain rates to 1.0 sec⁻¹. However, due to the limited number of irradiated specimens the majority of testing was done at two temperatures: 232°C and at the irradiation temperature. A temperature of 232°C was employed in this evaluation because it represents the approximate refueling temperature of LMFBR's and because Inconel 718 is used in refueling equipment. Results from tensile tests conducted at a conventional strain rate on irradiated material are presented in Table III and results of the high strain tests in Table IV. Included in these tables are total and fast fluences for each of the test specimens.

The temperature dependence of the tensile properties of irradiated Inconel 718 has been reported previously ⁽⁶⁾. Since the specimen material and irradiated experiment were identical to those of the present study, the

TABLE III

TENSILE PROPERTIES OF IRRADIATED INCONEL 718
(Annealed 1750°F/0.5 Hr. - Aged 1325°F/8 Hr. + 1150°F/10 Hr.)

Irrad. Temp. °C	Fluence- Total	10 ²² n/cm ² E>0.1 MeV	Test Temp. °C	ε- Sec-1	Strength - PEL 0.2%	· MPa (Ksi Yield) Ultin		ongati Total	on-% Unif.	Reduction of Area-%	Site Sequence
391	0.93	0.80	22 232 427 427 538 649	3x10 ⁻⁵	791 (114.7) 1153 803 (116.4) 1017 818 (118.7) 1006 695 (100.8 942 803 (116.4) 1009	(167.2) (147.5) (145.9) (136.6) (146.3)	1300 1153 1107 1121 1118	(188.5) (167.2) (160.6) (162.6) (162.2)	10.3 10.5 8.4 8.4 4.1 0.7	6.4 6.9 4.0 5.7 2.7	27.9 32.8 26.8 26.8 11.5 2.5	H0 4804 4805 4806 4824 4807
404	0.55	0.41	22 232 427 538 649	3x10 ⁻⁵	796 (115.4) 1027 859 (124.6) 1021 661 (95.9) 897 616 (89.4) 884 622 (90.2) 797	(149.0) (148.1) (130.1) (128.2) (115.6)	1245 1255 1107 1071 879	(180.5) (182.0) (160.6) (155.3) (127.5)	17.2 17.8 19.3 12.8 6.1	12.8 14.4 14.3 11.2 4.7	35.8 32.8 26.8 17.1 11.5	4808 4809 4810 4811 4861
593	0.86 0.86 0.98 0.98 0.98 0.86 0.94 0.98	0.71 0.71 0.86 0.85 0.85 0.71 0.81 0.86 0.71	22 232 232 427 538 593 593 593 649	3×10 ⁻⁵	656 (95.1) 910 610 (88.5) 814 616 (89.4) 863 701 (101.6) 854 644 (93.4) 814 587 (85.2) 769 678 (98.4) 832 599 (86.9) 812 554 (80.3) 738	(132.0) (118.0) (125.2) (123.8) (118.0) (111.5) (120.7) (117.8) (107.1)	1305 1160 1208 1136 1086 968 1039 1027 825	(189.3) (168.2) (175.2) (164.7) (157.3) (140.4) (150.7) (149.0) (119.7)	24.5 24.4 21.7 21.4 18.1 9.0 7.7 9.4 4.4	21.3 22.3 18.9 17.4 17.0 8.1 7.4 8.7 3.2	38.5 30.3 33.3 27.9 22.1 18.0 13.1 9.0 7.3	4818 4819 4822 4851 4852 4820 4850 4823 4821
593	1.78 2.44 2.44 1.78 2.44 1.78	1.30 1.96 1.96 1.30 1,96 1.30	22 232 427 538 593 649	3x10 ⁻⁵	628 (91.1) 914 718 (104.1) 912 746 (108.2) 893 684 (99.2) 841 658 (95.5) 877 656 (95.1) 791	(132.5) (132.3) (129.5) (122.0) (127.2) (114.7)	1281 1232 1140 1080 1029 838	(185.8) (178.7) (166.8) (156.7) (149.2) (121.6)	21.2 20.3 18.1 14.6 5.5 2,5	19.5 18.7 15.8 13.2 5.0 2,3	29.3 25.2 22.1 17.1 1.6 0.8	4843 4841 4915 4857 4838 4840
649	3.13 3.13 3.13 3.26 3.13 3.34 3.42	2.65 2.65 2.65 2.77 2.65 2.89 2.98	22 232 427 538 649 649 649	3x10 ⁻⁵	712 (103.3) 996 718 (104.1) 910 684 (99.2) 946 650 (94.3) 866 847 (122.9) 877 684 (99.2) 850	(144,4) (132,0) (137,0) (125,6) (127,2) (123,3)	1274 1232 1191 945 877 850	(184.8) (178.7) (172.8) (137.1) (127.2) (123.3)	17.0 20.3 16.0 3.1 1.7 0.3 0.1	12.7 18.7 14.3 3.1 1.6 0.3 0.1	28.7 25.2 19.5 4.1 3.2 2.4 0.8	4844 4842 4839 4914 4913 4912

TABLE IV
HIGH STRAIN RATE TENSILE PROPERTIES OF IRRADIATED INCONEL 718

Irrad.				arameters									
Temp.	Fluence- Total	10 ²² n/cm ² E>0.1 MeV	Temp. °C	έ -Sec ⁻¹		ength D.2%	- MPa (K Yield			Elongat Total	ion-% Unif	Reduction of Area-%	Site Sequence
396	0.86 0.86 0.86	0.71 0.71 0.71	232 427 649	7.4x10 ⁻³	803 (116.4) 780 (113.1) 960 (139.3)	1014 982 982	(147.0) (142.5) (142.4)	1170 1081 1001	(169.7) (156.8) (145.2)	7.2 5.7 3.9	5.9 4.4 3.2	26.2 28.7 14.7	НО 4847 4848 4860
396	0.86 0.93 0.86	0.71 0.80 0.71	232 427 649	1.0		1135 1051 989	(164.6) (152.4) (143.4)	1253 1135 1011	(181.7) (164.6) (146.7)		5.7 3.6 3.8	30.9 18.7 27.9	4905 4904 4906
593	0.94 0.94 0.96	0.81 0.81 0.83	232 593 649	7.4x10 ⁻³	667 (96.7) 622 (90.2) 650 (94.3)	845 801 823	(122.5) (116.2) (119.3)	1164 1038 1044	(168,8) (150.6) (151,4)	13.2	15.6 13.2 10.2	27.9 22.1 18.7	4849 4846 4845
593	0.98 0.97 0.98 0.97	0.86 0.84 0.86 0.84	232 538 593 649	1.0	840 (121.9) 729 (105.7) 791 (114.7) 803 (116.4)	950 827 859 851	(137.8) (120.0) (124.6) (123.4)	1228 1074 1057 1037	(178.1) (155.7) (153.3) (150.4)	18.0	14.1 13.3 16.2 12.0	41.3 24.6 24.6 18.9	4894 4897 4895 4896
593	1.48 2.44 1.48	0.99 1.96 0.99	232 593 649	7,4x10 ⁻³	757 (109.8) 791 (114.7) 672 (97.5)	900 904 801	(130.5) (131.1) (116.2)	1219 1107 994	(176.8) (160.6) (144.2)	12.5	15.5 10.7 7.6	35.0 23.8 11.5	4858 4856 4859
593	1.36 1.48 1.25	0.88 0.99 0.80	232 593 649	1.0	820 (118.9) 780 (113.1) 603 (87.4)	897 853 729	(130.1) (127.3) (105.7)	1185 1011 989	(171.9) (146.7) (143.5)	14.2	16.9 10.6 19.0	39.0 32.0 26.8	4902 4901 4903
649	3.34 3.34	2.89 2.89	232 649	7.4x10 ⁻³	925 (134.1) 791 (114.7)	1087 955	(157.7) (138.5)	1289 1094	(181.0) (158.6)		9.1 4.8	22.8 13,1	4854 4853
649	3.34 3.42 3.42	2.89 2.98 2.98	232 538 649	1.0.	847 (122.9)	1108 989 1009	(160.7) (143.4) (146.3)	1274 1173 1152	(184.8) (170.1) (167.1)		9.4 8.1 8.0	28.7 27.9 13.3	4898 4900 4899

available data provided results for this study at the lowest strain rate of interest, $3 \times 10^{-5}~\text{sec}^{-1}$. As shown in Reference 6, the ultimate and 0.2% yield strengths were not greatly affected by neutron irradiation at test temperatures to 649°C. However, there was a consistent trend toward slightly lower 0.2% yield strength at each test temperature for specimens irradiated at temperatures of 593 and 649°C. Since Inconel 718 is generally considered metallurgically stable at these temperatures for the time under irradiation (slightly less than 4000 hours) $^{(10)}$, it is possible that irradiation has enhanced the thermal overaging process thus resulting in the slight softening.

Consistent with the reduced 0.2% yield strengths, the ductility of irradiated material was slightly higher than unirradiated material for test temperatures to 538° C (6). As temperature is increased above 538° C the ductility is severely reduced and approached 2 per cent and less. This decrease has been attributed to helium embritlement (6).

The effect of strain rate on the tensile properties of irradiated Inconel 718 is illustrated in Figures 10-14. In each figure, data are presented for test temperatures of 232, 538, 593, and 649°C. Data from unirradiated materials (closed symbols) are included for comparison. The effect of strain rate on the ultimate strength is illustrated in Figure 10. For all irradiation and test conditions, the strength of irradiated material is very similar to that of the unirradiated material and the strain rate dependencies are nearly identical. Similar behavior is observed for the 0.2% yield strength as shown in Figure 11.

The strain rate dependence of the total and uniform elongation and reduction of area of irradiated material is presented in Figures 12, 13 and 14, respectively. For all test temperatures presented, the strain rate dependence of irradiated material approximates that for unirradiated material. Of significance is the fact that for test temperatures of 593 and 649°C, all ductility parameters increase with increasing strain rate.

To illustrate the effect of strain rate on the tensile properties of irradiated material, results from tests where the irradiation and test temperature were essentially identical are presented in Figure 15. Data are

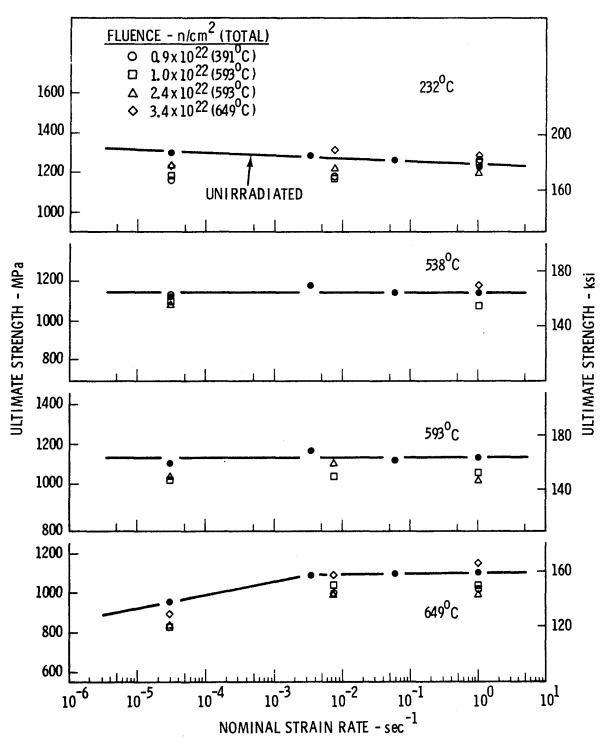


FIGURE 10. Strain Rate Dependence of the Ultimate Strength of Irradiated Inconel 718.

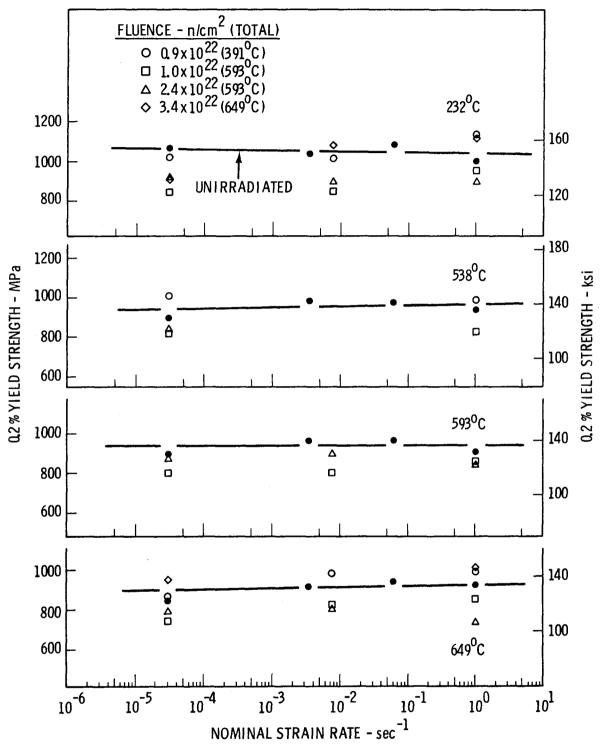


FIGURE 11. Strain Rate Dependence of the 0.2% Yield Strength of Irradiated Inconel 718.

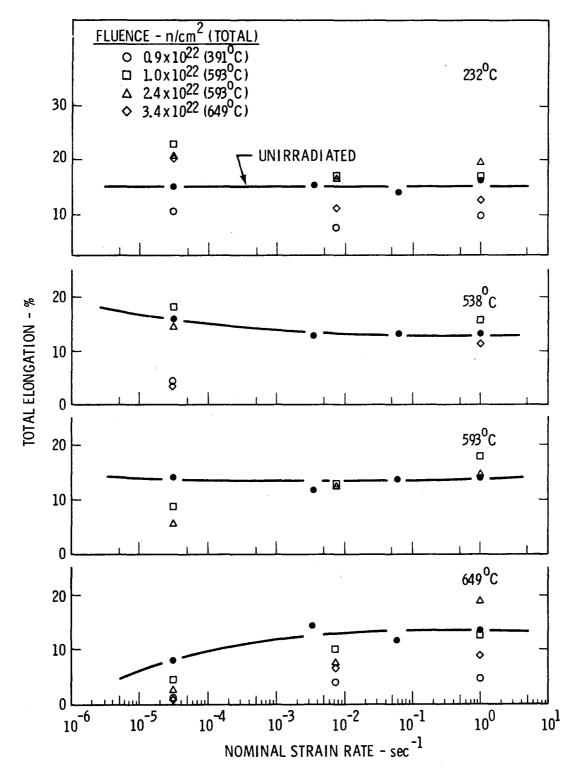


FIGURE 12. Strain Rate Dependence of the Total Elongation of Irradiated Inconel 718.

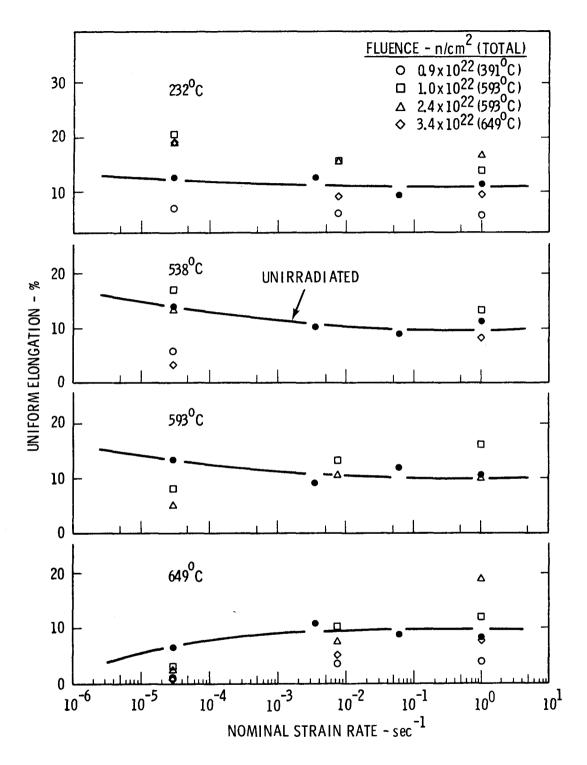


FIGURE 13. Strain Rate Dependence of the Uniform Elongation of Irradiated Incomel 718.

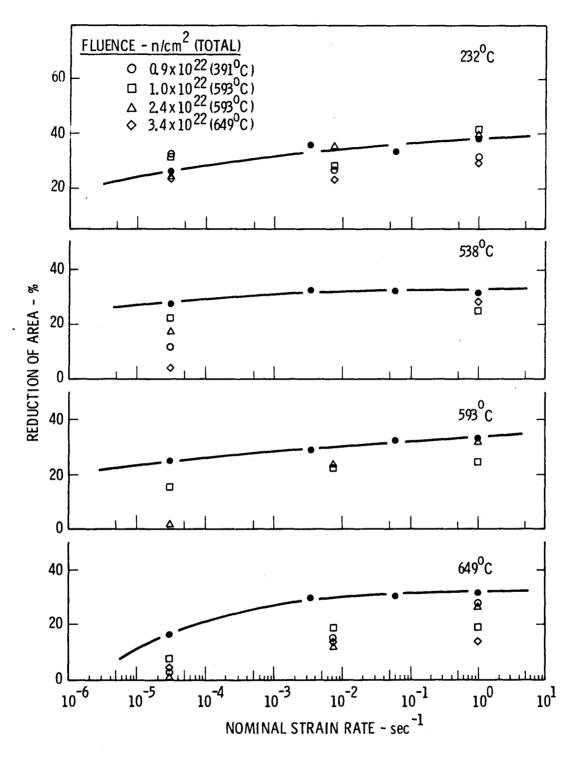


FIGURE 14. Strain Rate Dependence of the Reduction of Area of Irradiated Inconel 718.

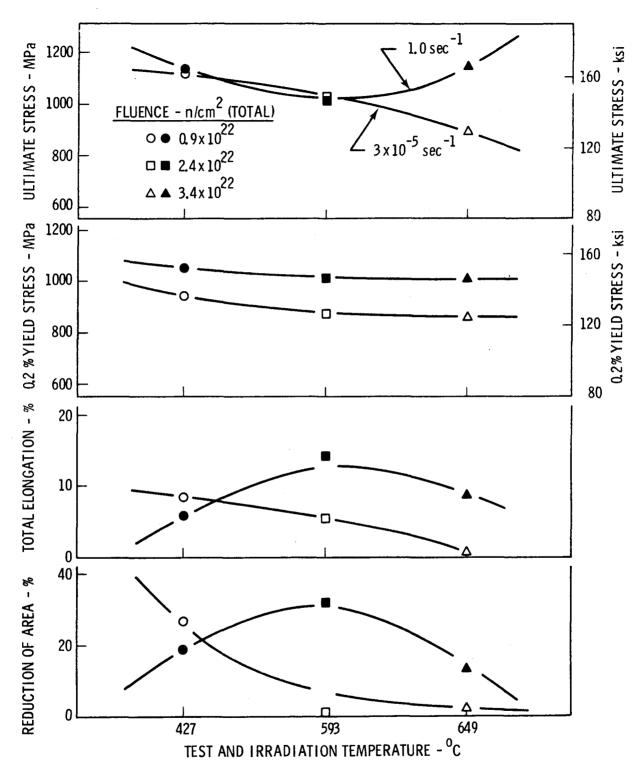


FIGURE 15. Tensile Properties of Irradiated Inconel 718 for Tests at the Irradiation Temperature.

included for two strain rates, 3×10^{-5} and $1.0 \, \mathrm{sec^{-1}}$. For irradiation and test temperatures of 427 and 593°C the ultimate strengths are identical between the two rates. At 649°C the ultimate strength is significantly higher at the higher strain rate. For each temperature the 0.2% yield strength is increased by increasing strain rate. Total elongation and reduction of area are decreased slightly at the highest strain rate at 427°C and significantly increased at the same strain rate at temperatures of 593 and 649°C. These trends are comparable with those of unirradiated material (Figure 16).

At irradiation temperatures of ~ 400 and 593°C specimens were exposed to different fluence levels thus allowing a comparison of strain rate effects at various fluences. For the lower irradiation (400° C) and test temperature (427° C) the ultimate strength appeared to decrease slightly with increasing fluence and no effect of strain rate was observed (Figure 17). The 0.2% yield strength slightly increased with fluence and increasing strain rate. Total elongation and reduction of area decreased with increasing fluence. The effect of increasing strain rate on the ductility properties was to decrease total elongation with little effect on reduction of area.

At an irradiation and test temperature of 593°C, little effect of increasing fluence or strain rate on strength properties was observed (Figure 18). Conversely, significant effects of both strain rate and fluence on total elongation and reduction of area were observed. At the lowest strain rate, both ductility properties decreased significantly with increasing fluence. Increasing strain rate to 1.0 sec⁻¹ resulted in ductility of the irradiated material nearly identical to that of unirradiated material. Although uniform elongation data are not presented in this figure or Figure 17, the trends are consistent with total elongation behavior.

Since the strain rate dependence of the tensile properties of irradiated material was essentially identical to that for the unirradiated material, it appears that irradiation does not greatly alter the mechanisms controlling deformation in Inconel 718. Evidence to support this fact was obtained from the autographic records of the tests on irradiated materials.

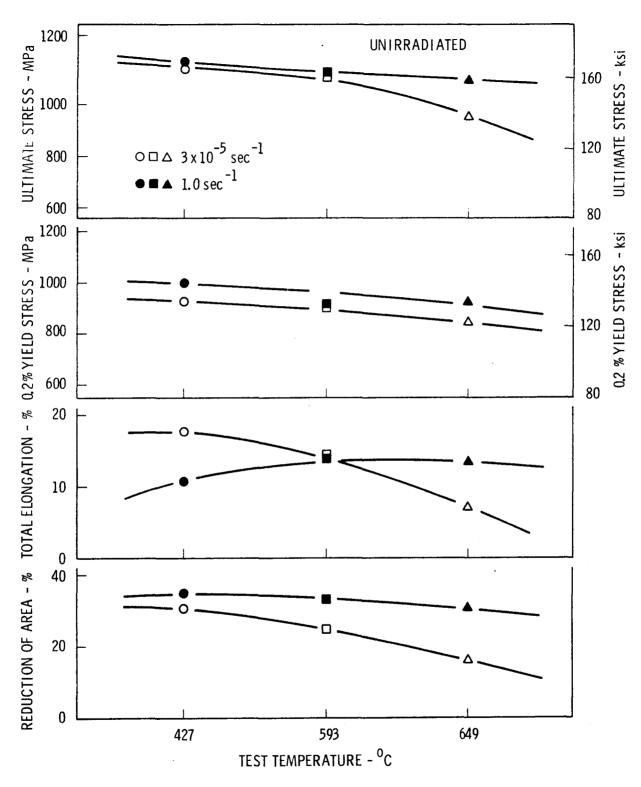


FIGURE 16. Tensile Properties of Unirradiated Inconel 718 at Temperatures of 427, 593, and 649°C.

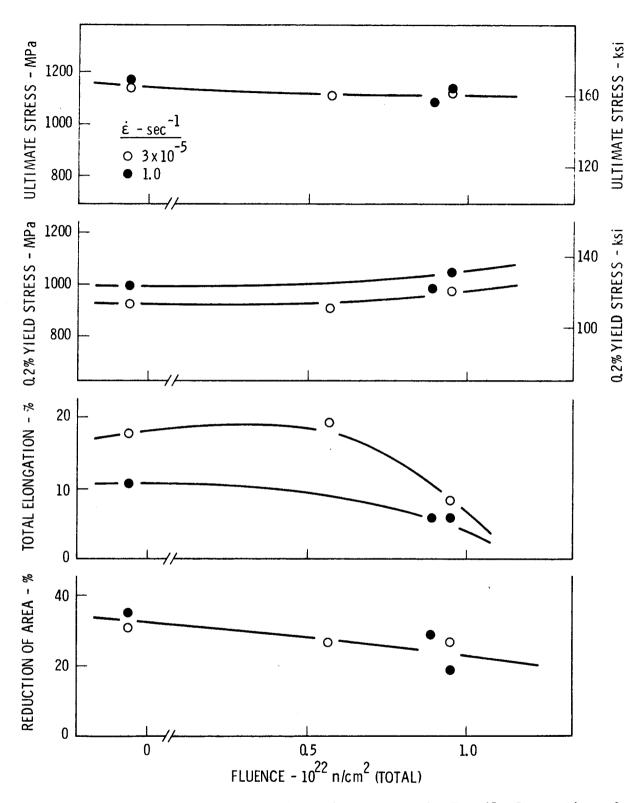


FIGURE 17. Effect of Fluence and Strain Rate on the Tensile Properties of Irradiated Inconel 718 -- Irradiated at \sim 400°C and Tested at 427°C.

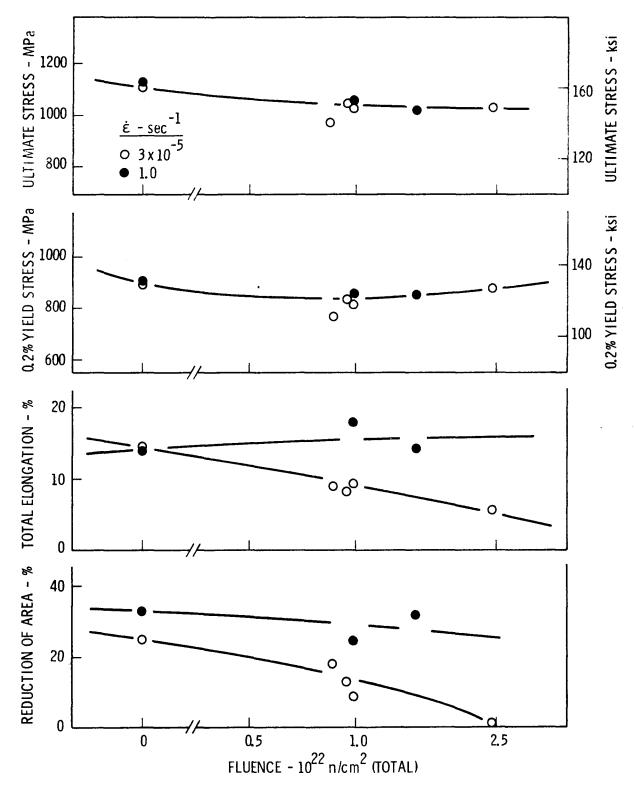


FIGURE 18. Effect of Fluence and Strain Rate on the Tensile Properties of Irradiated Inconel 718 -- Irradiated and Tested at 593°C.

Serrations were observed at the various test temperatures with similar shapes and magnitudes as compared to those in Figure 7. Also, increasing strain rate resulted in behavior of irradiated material which was consistent with that presented in Figure 8 for unirradiated material.

SUMMARY AND CONCLUSIONS

In the present study the effect of strain rate on the elevated temperature tensile properties of unirradiated and irradiated Inconel 718 were determined. Irradiation conditions included total neutron fluences to 1.0 x 10^{22} n/cm² at 400°C, to 2.4 x 10^{22} n/cm² at 593°C, and 3.4 x 10^{22} n/cm² at 649°C. Test parameters included temperatures to 649°C and strain rates to 1.0 sec⁻¹. The following conclusions can be made regarding the results of this study:

- The strength and ductility of unirradiated Inconel 718 are insensitive to strain rate for temperatures to 538°C. At temperatures above 538°C both properties significantly increased with strain rate.
- Neutron irradiation at $\sim 400^{\circ}\text{C}$ did not alter this behavior since postirradiation properties were essentially unaffected by increasing strain rate.
- At irradiation temperatures of 593°C and above, strength properties were essentially consistent with unirradiated behavior.
 Ductility parameters were unaffected by strain rate for temperatures to 538°C. At higher temperatures ductility increased significantly with strain rate and approached unirradiated values.
- The severe ductility losses observed in all irradiated specimens at the lowest test rate $(3 \times 10^{-5} \text{ sec}^{-1})$ and higher test temperatures (593°C and above) is believed to result from helium embrittlement.

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