# Radiation Shielding Using Depleted Uranium Oxide in Nonmetallic Matrices

**July 2002** 

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# RADIATION SHIELDING USING DEPLETED URANIUM OXIDE IN NONMETALLIC MATRICES

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# ABBREVIATIONS AND ACRONYMS

ASTM American Society for Testing and Materials

BNL Brookhaven National Laboratory

BWR boiling water reactor DOE U.S. Department of Energy

DU depleted uranium

DUAGG depleted uranium aggregate
DUCRETE depleted uranium concrete
DUPoly depleted uranium oxide powder

EDX X-ray spectroscopy

GNB Gesellschaft für Nuklear-Behälter

HLW high-level-waste

IAEA International Atomic Energy Agency

ICP ion-coupled plasma

INEEL Idaho National Engineering and Environmental Laboratory

LWR light-water reactors

ORNL Oak Ridge National Laboratory PYRUC Pryolytic uranium compound

RBMK Russian graphite moderated boiling water reactor

RCRA Resource Conservation and Recovery Act

SEM scanning electron microscope

SNF spent nuclear fuel

TCLP Toxic Characteristic Leaching Procedure

tonne metric ton

TRU transuranic waste UO<sub>x</sub> uranium oxide XRD X-ray diffraction

YMP Yucca Mountain Project

### **ABSTRACT**

The depleted uranium (DU) inventory in the United States exceeds 500,000 metric tons (tonnes). This report reviews the status of U.S. Department of Energy (DOE) research concerning the use of its inventory of DU as neutron and gamma shielding. The focus is the use of uranium oxide (UO<sub>x</sub>) in nonmetallic matrices; this report does not discuss the ongoing research on CerMets, which incorporate UO<sub>x</sub> in metallic matrices, which is discussed elsewhere.1 This program envisions that a large portion of the U.S. inventory of DU will be used in the fabrication of nuclear shielding for the storage, transport, and disposal of spent nuclear fuels. Just in the storage of commercial U.S. spent nuclear fuel (SNF), the cumulative amount of DU oxide (DUO<sub>2</sub>) that could be used in dry-storage casks though 2020 is 408,455 tonnes (360,058 tones of DU).

The purpose of this U.S. research effort is to develop DU shielding technologies to the point that a demonstrated technical basis exists for deployment. In particular, a need exists to (a) establish the ability to manufacture DU aggregates in large, heavy concrete shapes; (b) optimize the design and costs; and (c) ensure confidence in the reliability and safety of the chemical and physical stability of the DU aggregate. In addition, the neutron-shielding characteristics of these aggregates and their binders must be enhanced. This research and development will underpin public, regulatory, and purchaser confidence.

Current concepts are to form UO<sub>x</sub> aggregates and to combine these aggregates with binders that enhance neutron shielding. Uranium is a very effective gamma shield because of its high density and high Z number. Binders under consideration include cementitious pastes, thermoplastic polymers, and pyrolytic carbon. The use of these DU-based shielding materials will greatly reduce the size and weight of storage, transport, and disposal casks. The economic advantage gained through using smaller and lighter casks will offset the increased fabrication costs of using the DU.

In 1993, the DOE Office of Environmental Management began investigating the potential use of DU in heavy concretes. DUCRETE<sup>TM</sup>, or depleted uranium concrete, was an outcome of that investigation. This material is a high-density concrete that uses depleted uranium ceramic aggregates. Researchers also developed DUPoly, which uses depleted uranium oxide powder as the filler material in a thermoplastic polyethylene binder material to produce a high-density shielding material. A third concept uses a micro-aggregate DU oxide (DUO<sub>2</sub>) that is produced by a sol-gel precipitation of uranium into microspheres in a process developed for nuclear fuel technology in the late 1960s. These sol-gel particles are then mixed with an organic binder and pyrolyzed to make uranium carbide and/or  $UO_x$  pyrolytic-carbon matrices. This matrix is called PYRUC.

All of these materials use DU in a neutron-moderating or -absorbing binder. In a dense, compact shield, these materials have both an efficient gamma absorber and a neutron-slowing material. Therefore, they are ideal for shielding sources, such as spent nuclear fuels and vitrified high-level waste. This report provides the background for these technologies and some of the associated mechanical and nuclear performance data that have been developed.

<sup>1.</sup> C.W. Forsberg, L.B. Shappert, P. Byrne, and B. Broadhead, "CerMet Transport, Storage, and Disposal Packages Using Depleted Uranium Dioxide and Steel," *Proceedings of the 13th International Symposium on the Packaging and Transport of Radioactive Materials for the Institute of Nuclear Materials Management*, Chicago, Ill., September 7, 2001 (Warrendale, Pa.: Materials Research Society, August 2002).

#### 1. INTRODUCTION

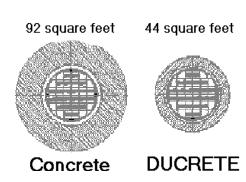
One of the most obvious uses for the large U.S. inventory of depleted uranium (DU)—which exceeds 500,000 metric tons (tonnes)—is as nuclear shielding. A research program being conducted by the U.S. Department of Energy (DOE) envisions that a large portion of this DU will be used in the fabrication of nuclear shielding for the storage, transport, and disposal of spent nuclear fuel (SNF). DU metal has been used in casks as shielding because its high density provides the needed gamma attenuation for the lowest-weight and smallest casks. Studies have assessed the use of uranium metal for shielding in both spent fuel and high-level waste (HLW) casks. A review of DU metal production and fabrication costs showed that depleted metal was more expensive than other common shielding materials such as steel, lead, and concrete. Therefore, the primary application for uranium metal shielding is for transportation casks, where the most stringent total-package size and weight limits exist and where high-cost, DU metal shielding can be justified. Also, there is an added benefit to the nuclear community if this use as shielding consumes large quantities of DU from the existing national inventory.

These findings led to the consideration of alternative uses for DU such as a DU ceramic, which is still very dense but has considerably lower production and fabrication costs than DU metal. The first alternative developed was a concrete called DUCRETE<sup>TM</sup>, which was followed by DUPoly and PYRUC. DUPoly uses depleted uranium oxide powder as the filler material in a thermoplastic polyethylene binder material to produce a high-density shielding material. PYRUC uses a microaggregate DU oxide (DUO<sub>2</sub>) that is produced by a sol-gel precipitation of uranium into microspheres in a process developed for nuclear fuel technology in the late 1960s. These sol-gel particles are then mixed with an organic binder and pyrolyzed to

make uranium carbide and/or UO<sub>x</sub> pyrolyticcarbon matrices.

All of these concepts have in common the use of DU in a neutron-absorbing binder. This provides a material that has characteristics of both an efficient gamma absorber (uranium) and a low-Z neutron-slowing material (hydrogen, carbon. etc.). Figure 1 shows the effectiveness of using DUO<sub>2</sub>, such as DUCRETE, to reduce the size and weight of a drystorage cask or silo for spent nuclear fuel.

This report discusses the backgrounds of these technologies and some of the mechanical and nuclear performance data developed. It also identifies additional development requirements.



**Fig. 1. Comparative diameters of concrete and DUCRETE dry-storage cask or silo.** Using DUCRETE in a spent nuclear fuel cask or silo reduces the weight by 30%, the footprint by 50%, and the diameter from 132 in. (3.5 m) to 90 in. (2.3 m).

# 2. DUCRETE CONCRETE

DUCRETE consists of a DU ceramic that replaces the coarse aggregate used in standard concrete. The DU coarse aggregate is combined with Portland cement, sand, and water in the same volumetric ratios used for ordinary concrete. If the ceramic can be produced at a low enough cost, it would be practical to consider using DUCRETE concrete as a shielding material. The cost of concrete cask fabrication is low when compared to fabricating steel, lead, and DU metal casks.

# 2.1 Early Development

DUCRETE concrete was conceived at the Idaho National Engineering and Environmental Laboratory (INEEL) by W. Quapp and P. Lessing, who jointly developed the process and were awarded both U.S. and foreign patents in 1998 and 2000, respectively. DUCRETE is formed by agglomerating DUO<sub>2</sub> to a dense, stable, low-cost aggregate and combining it with the normal concrete paste (cement, sand, and water). This matrix has both high-Z materials for gamma attenuation and low-Z material for neutron attenuation. Emulating nuclear fuel technology, the sintered uranium oxide (UO<sub>x</sub>) aggregate has a very high density (>95% theoretical density). Thus, a theoretical concrete density of 7.2 g/cm<sup>3</sup> is possible.

Based on this conceptual work, J. Sterbentz of INEEL performed the first shielding calculations. <sup>6</sup> Initial shielding evaluations were made for DUCRETE shielding in a spent fuel application. Figure 2 shows the nuclear shielding effectiveness of this conceptual DUCRETE shielding material. The figure shows comparisons of the relative effectiveness for gamma and neutron attenuation of DUCRETE and that of other common shielding materials in a proposed SNF storage silo or cask.

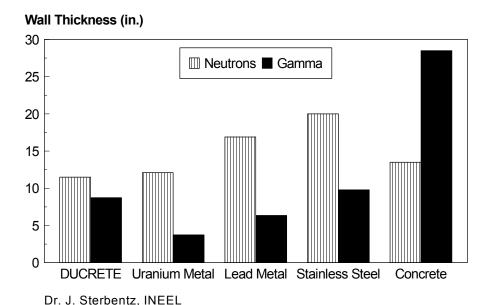
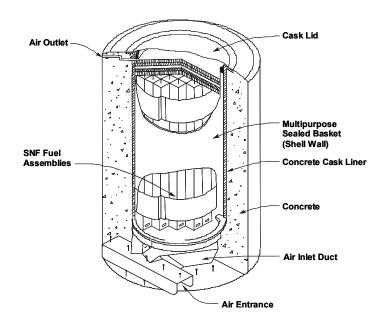


Fig. 2. Comparison of storage cask or silo wall thicknesses required to attenuate neutron and gamma doses from 24 pressurized water reactor (PWR) spent nuclear fuel assemblies to 10 mR/h.

Another series of studies conducted by two firms in the U.S. private sector, Packaging Technology and Sierra Nuclear Company, provided a commercial perspective for the DUCRETE concept. <sup>7,8,9,10</sup> Their results showed that if DUCRETE is found to be producible at relatively low cost, the concept appeared to be technically sound.

Sierra Nuclear Company found that spent fuel storage casks, with performance improvements and weight savings (Fig. 3), would nearly offset the greater fabrication costs through eliminating redundant handling costs. Sierra Nuclear also showed that an optimum uranium-to-binder ratio exists for a combined attenuation of gamma and neutron radiation at a given wall thickness (Fig. 4). A balance needs to be established



**Fig. 3. Diagram of the Sierra Nuclear VSC-24 spent fuel storage cask.** Substituting DUCRETE into this current design reduced the weight of the cask (24-PWR assemblies) by 30 tons and the footprint by a factor of 2 (see Fig. 1).

between the attenuation of the gamma flux in the  $UO_x$  and the cement phase with water to attenuate the neutron flux. Figure 4 shows the DU aggregate-loading-to-attenuation relationships for two different types of sand added to the DUCRETE binder materials: ordinary quartz sand (SiO<sub>2</sub>: density of 2.65 g/cm³) and colemanite sand (Ca<sub>2</sub>B<sub>6</sub>O<sub>11</sub>•H<sub>2</sub>O: density of 2.4 g/cm³). With regard to neutron attenuation, the colemanite sand has the added advantage of adding boron to the shield matrix, for a small reduction in overall density. Neutron-absorbing elements like hafnium and gadolinium also may be used to enhance the shield's neutron attenuation, allowing even thinner, lighter combinations of  $UO_x$  and binders.

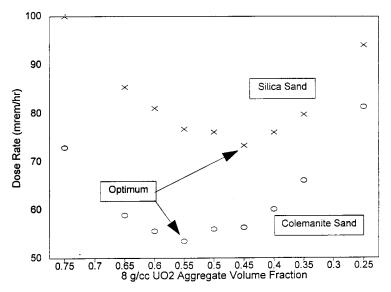


Fig. 4. Cask external dose vs DUCRETE aggregate volume fraction (Ref. 9): optimum aggregate loading.

# 2.2 DUCRETE Proof-of-Concept Testing

Lessing conducted proof-of-concept tests to determine whether there were deleterious chemical reactions between depleted urania aggregate ( $UO_2$ ) and the concrete paste. <sup>11</sup> Under aqueous oxidizing conditions in the cement paste,  $U_3O_8$  is the more thermodynamically stable form of uranium. The transition from  $UO_2$  to  $U_3O_8$  results in an expansion of the aggregate that could generate stresses that could crack this shield material and lower its compressive strength. However,  $U_3O_8$  DUCRETE is less dense than  $UO_2$  DUCRETE.

Initial tests used DU sintered fuel pellets as simulated coarse aggregates. These fuel pellets were right circular cylinders 0.25 in. (0.635 cm) in diameter by 0.5 in. (1.25 cm) long, with smooth surfaces. DUCRETE concrete samples were made with these fuel pellets, and blank concrete samples were made using ordinary gravel. The volume densities of all these samples were varied to determine the maximum loading while still retaining acceptable mixing characteristics and the compressive strength of the concrete. A series of DUCRETE and normal concrete control samples were prepared and cured for 7, 28, and 90 days and compression-tested to get reference data. The results are shown in Fig. 5. A sample size of 2 in. (5 cm) in diameter by 4 in. (10 cm) long was selected to minimize the use of the uranium ceramic per sample. These small specimen sizes are responsible for the large spread in the data for both materials. For the size of aggregate used, the cylinders should be at least 3 in. (7.6 cm) in diameter by 6 in. (15.2 cm) long. The normal effects of cure time on the compressive strength are

masked by the data spread. Nevertheless, it can be seen that the compressive strength of DUCRETE and normal concrete seems comparable.

To evaluate the potential for any chemical interaction between the UO<sub>x</sub> ceramic and the cement phase, additional samples were prepared and cured for 7, 14, and 28 days at temperatures ranging from room temperature to 250°C. After the appropriate curing period, the DUCRETE samples were compression-tested

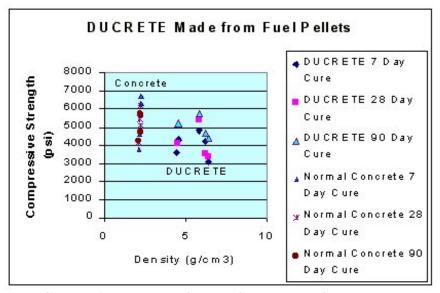


Fig. 5. Compression test results from DUCRETE and reference concrete samples (room temperature).

to compare their strengths with that of ordinary concrete with the same volume fraction of course aggregate. These tests assessed the effects of elevated temperature on the samples' compressive strengths. The data from this test are shown in Fig. 6. Lessing found no discernable difference in the strength for samples cured at temperatures below 150°C as compared to samples cured at room temperature, within the statistical uncertainty of the data. For the samples cured at elevated temperatures (250°C), damage occurred to both normal and DUCRETE samples. However, some DUCRETE samples had crumbled at the end of 14 days, implying a different damage mechanism than for the normal concrete.

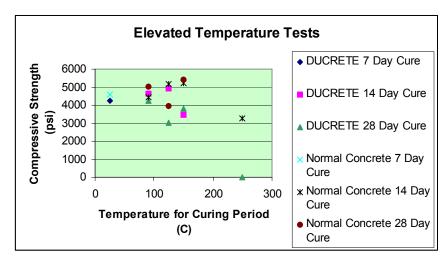


Fig. 6. Compression test data following elevated-temperature exposures.

Lessing investigated the damage mechanism. 12 He concluded that while there did not appear to be any chemical reaction between the concrete materials and the fuel pellets, some oxidation of the UO<sub>x</sub> pellets occurred during the high-temperature exposure tests. The oxidation, although invisible to the unaided eye, was apparently sufficient when com-

bined with the concrete dehydration at the test conditions to cause dramatic failure of the DUCRETE samples due to volume increase of the coarse aggregate during the transition from  $UO_2$  to  $U_3O_8$ .

While UO<sub>2</sub> was expected to behave as a reactive coarse aggregate, the temperature threshold at which reactions occur and the reaction rates are still not known. Under dry conditions at elevated temperature, concrete can dehydrate. For most storage and disposal casks and conditions, the bulk-concrete shield temperature is limited to 66°C. But localized temperatures at the inner walls of the shielding casks can be up to 149°C. During a short-term accident, some parts of the shield could reach temperatures of up to 343°C. Table 1 shows the specifications from several cask vendors. <sup>12, 13</sup> The performance limits shown in the table are derived from the American Concrete Institute standards.

Table 1. Concrete temperature limits for nuclear fuel storage casks

Source	References cited	Normal "bulk" temp.	"Local" long-term temp.	Short-term (accident) temp.
Depleted Uranium Concrete Container Feasibility Study	ACI-349 Appendix A and NRC Guidance	66°C (150°F)	93–149°C (200–300°F)	177–343°C (350–650°F)
Pacific Nuclear "NUHOMS" System <sup>a</sup>	ACI-349 Appendix A	66°C (150°F)	93°C (200°F)	177°C (surface) to 343°C (local) (350–650°F)
Sierra Nuclear "VSC" Systems <sup>b</sup>	ACI-349 Appendix A and NRC Guidance	66°C (150°F)	149°C (local) (300°F)	_
Babcock & Wilcox Fuel Company <sup>c</sup>	_	_	121°C (250°F)	

<sup>&</sup>lt;sup>a</sup>R. T. Haelsig, *Depleted Uranium Concrete Container Feasibility Study*, EGG/MS-11400 (Washington, D.C.: Packaging Technology Inc., 1994).

<sup>&</sup>lt;sup>b</sup>Pacific Sierra Nuclear Associates, *Safety Analysis Report for the Ventilated Storage Cask System*, PSN-91-001 Rev. 0 (Scotts Valley, Calif., October 1991).

<sup>&</sup>lt;sup>c</sup>Babcock & Wilcox Company, *BR-100 Shipping Cask Preliminary Design Report*, 51-1177082-01 (Lynchburg, Va., n.d.), p. II-3-11.

At the higher temperatures, the oxidation of the fuel pellets may have been accelerated, but the dehydration of the concrete paste could also have contributed to the disintegration. The dominant mechanism is not yet known. Nevertheless, it was concluded that if the  $UO_x$  aggregate could be rendered less reactive with water and oxygen at elevated temperatures, DUCRETE should be more suitable for cask applications.

# 2.3 Development of DUAGG

"Depleted uranium aggregate" (DUAGG) is the term applied to the stabilized, depleted  $UO_x$  aggregate that was developed to reduce the rates of  $UO_2$  oxidation previously described. In the formation of DUAGG, a coating covers the surfaces of sintered urania particles and fills between the grains. This coating is an oxygen barrier. The sintering temperature normally associated with  $UO_x$  sintering (~1700°C) is also reduced to below 1300°C, an added benefit from a manufacturing perspective. Other work at INEEL led Lessing to consider a basalt-based binder because of its demonstrated resistance to corrosion in hot, aqueous environments. Therefore, DUAGG was formulated with inorganic binder materials consisting of clays, boria, iron oxide, and other materials similar in composition to basalt. Basalt is a dense crystalline rock of volcanic origin, composed largely of plageioclase feldspars ([Na, Ca] Al [Si, Al]Si<sub>2</sub>O<sub>8</sub>) and dark minerals such as pyroxene (~n[Si<sub>2</sub>O<sub>6</sub>]~) and olivine ([Mg, Fe<sub>2</sub>] SiO<sub>4</sub>). A process flow diagram for the INEEL manufacturing process is shown in Fig. 7.

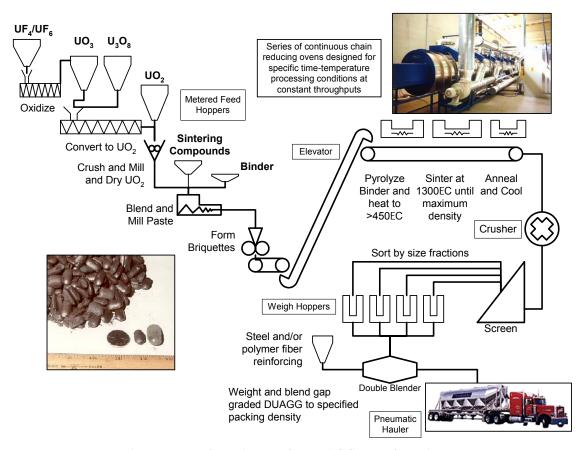


Fig. 7. Process flow diagram for DUAGG manufacturing.

Lessing designed the mixture so that the aggregate could be made using liquid-phase sintering at reduced temperatures. Sintering temperatures of 1250°C to 1300°C produced an aggregate exceeding 95% of theoretical density. Since the mixture was about 7 wt % of lower density oxides and only 93% urania (UO<sub>2</sub>), the resultant theoretical density was reduced to about 8.9 g/cm³, compared to 10.4 g/cm³ for 95% dense urania. Based on the work of Hopf,<sup>8</sup> this density loss results in only a slight increase in the wall thickness for the same attenuation and external dose. Research on the optimization of the time-temperature processing conditions, as well as on the introduction of neutron poisons into the DUAGG matrices, is continuing.

The challenge in this process is to obtain the proper sintering cycle to both reduce the  $UO_3$  or  $U_3O_8$  to  $UO_2$  and to come close to achieving the theoretical density. This requires a two-step sintering process: (1) achieving a hold period at 900°C for about 2 hours in a reducing atmosphere, followed by (2) completion of sintering at 1250 to 1300°C.

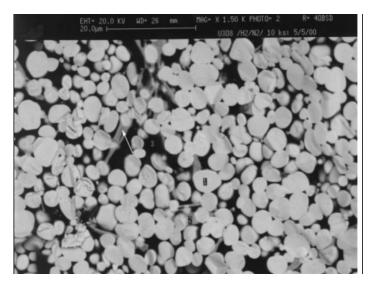


Fig. 8. Backscattered electron image of a fractured surface of DUAGG. The arrow highlights the U-Ti-oxide phase. The light, rounded particles are  $UO_2$ , while the black area surrounding the  $UO_2$  is the silicate phase, which is molten during sintering.

Based on optical examination of DUAGG, the liquid-phase binder seemed to coat all urania particles. A scanning electron microscope (SEM) photograph of a DUAGG sample is shown in Fig. 8. This figure shows the results from the liquid-phase sintering process. The spherical UO<sub>x</sub> grains are nearly completely surrounded by the silicate-based phase (the liquid phase at the sintering temperature). This coating gives corrosion and leach resistance to DUAGG.

The  $UO_x$  phase was determined by X-ray diffraction to be predominantly  $UO_2$ . The  $UO_2$  appears in all specimens analyzed as rounded particles measuring  $\sim 1-5~\mu m$ . The silicate phase wets the  $UO_x$  phase and fills

space between the UO<sub>x</sub> particles, demonstrating the success of the liquid-phase sintering. The silicate phase contains a mixture of Si, Al, Ca, Ti, K, Na, Zr, Fe, and Mg. Microchemical analysis by energy-dispersive X-ray spectroscopy (EDX) in the SEM revealed similar composition from point to point both within and between samples. This indicates that a homogeneous silicate phase exists. X-ray diffraction (XRD) showed the silicate phase to be predominantly crystalline.

Lessing repeated most of the previously described compression tests using DUCRETE made with DUAGG, focusing on the elevated temperature performance. These data are shown in Fig. 9. Again these tests have a large data spread caused by the use of small compression-test cylinders relative to the size of the coarse aggregate. In general, the compressive strength at low temperatures was similar to that obtained in prior test results. However, at the higher temperatures, the behavior of the DUAGG generally improved. Still at 250°C, the compressive strengths are reduced compared to the baseline data for both DUCRETE and normal concrete. Because of the scatter in the data, however, the

difference between the performance of ordinary concrete and DUCRETE was not statistically significant.

Both types of concrete seem to be affected by dehydration at 250°C. <sup>14</sup> However, under microscopic examination, no discernable effect of oxidation was observed on the DUAGG aggregate surfaces. Apparently, the liquid-phase sintering materials coat the UO<sub>2</sub> grains and improve the durability of the DUAGG, when compared to durability achieved using sintered UO<sub>2</sub> pellet aggregates, as in the first tests (Fig. 6).

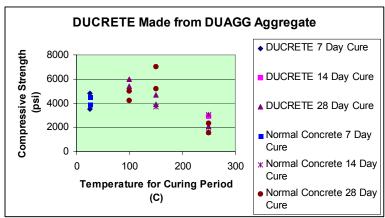


Fig. 9. Compression test results with DUCRETE exposed to high-temperature cure (DUAGG aggregate).

The compression test data in Fig. 9 is tabulated in Table 2. To compare the effect of DUAGG versus fuel pellets on compressive strength, the compressive strength data were plotted against the 28-day temperature data (Fig. 10). The DUCRETE samples made from DUAGG aggregate appear to have better strength at all curing temperatures. This could be caused by the poor statistical quality of the results or could be explained by the differing physical shape of the two aggregates. As previously noted, the fuel pellets were smooth right circular cylinders (0.25 in. diam by 0.5 in. long), whereas the DUAGG aggregate, beginning with sintered discs of 0.75 in. in diameter by 0.375 in. thick, was crushed and screened to sizes between 0.187 and 0.5 in. with rough surfaces. The INEEL work concluded at this point, and Starmet Corporation in South Carolina performed subsequent development.

Table 2. DUCRETE compressive strength after elevated temperature

Sample ID	Aggregate type	Aging temperature (°C)	Time (days)	Compressive strength (psi)	Average strength (psi)
OST-1	Gravel	Baseline	Baseline	3899	
OST-2	Gravel	Baseline	Baseline	4535	4217
ORT-1	DUAGG	Baseline	Baseline	3500	
ORT-2	DUAGG	Baseline	Baseline	4790	4145
OST-3	Gravel	100	28	5033	
OST-4	Gravel	100	28	4239	4636
ORT-3	DUAGG	100	28	6007	
ORT-4	DUAGG	100	28	5399	5703
OST-5	Gravel	150	14	3700	3700
OST-6	Gravel	150	28	5193	
OST-7	Gravel	150	28	6998	6096
ORT-5	DUAGG	150	28	4659	
ORT-6	DUAGG	150	28	3883	4271
OST-9	Gravel	250	14	1655	
OST-10	Gravel	250	14	3026	2341
ORT-8	DUAGG	250	14	2911	2911
OST-11	Gravel	250	28	2349	

Table 2. DUCRETE compressive strength after elevated temperature

Sample ID	Aggregate type	Aging temperature (°C)	Time (days)	Compressive strength (psi)	Average strength (psi)
OST-12	Gravel	250	28	1545	1947
ORT-9	DUAGG	250	28	2084	2084

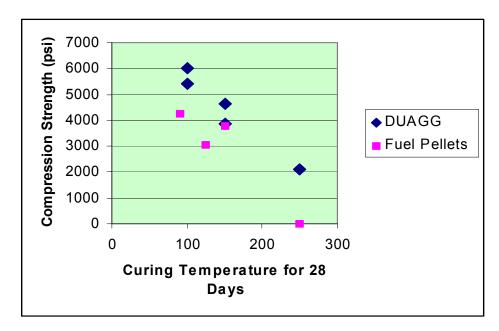


Fig. 10. Comparison of the compressive strengths between 28-day cured samples made from DUAGG and fuel pellet aggregates.

# 2.4 Commercial DUAGG Manufacturing

Starmet Corporation envisioned that the DUCRETE process could be commercialized if the production and fabrication costs that were predicted in the previous studies could be achieved. The UO<sub>3</sub>, stored at the Savannah River DOE site adjacent to the Starmet facility, could be used in DUCRETE casks for HLW storage needs at that site. Therefore, Starmet established a pilot-scale manufacturing facility in 1997 to demonstrate the manufacturing process developed by Lessing. Starmet used UO<sub>3</sub> as a starting oxide because it was the most abundant DU oxide material in the DOE inventory.

The Starmet process was originally designed to produce about 1000 lb/day of DUAGG. The only significant difference between the Starmet and Lessing processes (besides capacity) was Starmet's use of a rotary briquetting press (see Fig. 11) instead of the uniaxial laboratory press used at INEEL.

The first briquettes that Starmet produced were not of the proper density because of the limitations of the furnace. The furnace's two-zone temperature control did not allow the proper combination of time and temperature for the respective reduction and sintering steps. Consequently, the process steps were adjusted to achieve a satisfactory briquette density. Finally, the sintering was successfully completed using the furnace in two separate passes in which the reduction and sintering steps were accomplished

independently. The briquettes shown in Fig. 11 have a density of 8.1 g/cm<sup>3</sup>. Because of the furnace's limitations, Starmet was never able to accomplish the target production capacity for DUAGG.

Production was limited to about 100 lb/day. Nevertheless, it is still believed that a furnace with the appropriate temperature-zone control could achieve effective production of DUAGG of 1000 lb/day.

# 2.5 Mechanical Properties

Using different DUCRETE batch mixtures, Starmet performed further testing. Selected composition data are provided in Table 3, and the results of compression tests are presented in Table 4. Again, as with Lessing's data, the compression-testing results for these samples show a lot of variability. These tests used the same small (2 × 4 in.) cylinders used previously.



Fig. 11. DUAGG briquettes produced at Starmet.

Table 4 shows the effect on the density with samples made from lower-density DUAGG (8.1 vs 7 g/cm³). For comparison, normal concrete has a density of about 2.26 g/cm.³ Thus, even the "low-density" DUAGG formulation of DUCRETE results in much higher-density concretes than normal. A revised manufacturing process developed by Starmet produced an aggregate of 8.8 g/cm³, which makes concrete of about 6.3 g/cm³.

Table 3. Composition of DUCRETE samples and physical properties

Major ingredient <sup>a</sup>	Sample 7 ratios	Sample 8 ratios
Cement	1	1
DUAGG	9.6	8.18
Fly ash	0.2	_
Water	0.32	0.29
Cured density (g/cm <sup>3</sup> )	5.72	5.87

<sup>&</sup>lt;sup>a</sup>Both samples contained small quantities of thin metal steel fibers and superplastizer.

Table 4. DUCRETE density and compressive strength as a function of composition

Sample no.	Measured density (lb/ft³)	Strength, (psi)	DUAGG density <sup>a</sup> (g/cm <sup>3</sup> )	Small fines
6	5.66	5101	8.1	Fly ash
7	5.72	4310	8.1 <sup>b</sup>	Fly ash
8	5.87	4430	8.1 <sup>b</sup>	None
13	4.81	3880	7	Micro silica
14	4.73	4390	7	Micro silica

<sup>&</sup>lt;sup>a</sup>DUAGG briquettes are crushed and screened to yield American Concrete Institute no. 8 size fraction. <sup>b</sup>Sample contained about 0.36 wt % metal fibers to increase DUCRETE flexural strength.

# 2.6 Nuclear Properties

Most of the initial nuclear work done on DUCRETE has used attenuation calculations that assume various cask designs. <sup>6-9, 13, 14</sup> Experimental data were taken on two samples at the University of Missouri by Professor William Miller to determine the gamma attenuation. Figure 12 depicts the measured half-value thickness plotted against density for DUCRETE along with the textbook values

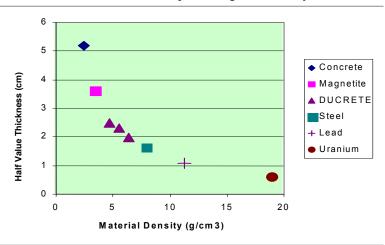


Fig. 12. Half-value layer thickness for DUCRETE shielding compared to other traditional shielding materials at 1.25 MeV.

for other materials.<sup>15</sup> The two lower DUCRETE densities are measured values, while the third and highest density is a predicted value based on aggregate density. The DUCRETE data were taken using <sup>60</sup>Co as the radiation source. <sup>16</sup>

Neutron attenuation is a function of the hydrogen content in the material, which in this case is the water bound in the cement paste and its pores. Holtec International Company made an independent neutron attenuation

measurement of a sample in March 2000 using a graphite-moderated reactor neutron source. Transmission measurements on a 2-in.-thick slab confirmed the expected attenuation.

# 2.7 Environmental Characteristics of DUCRETE

Potential leaching of DU into the environment is a consideration whether uranium forms the material for spent fuel storage casks, is used for other shielding applications, or is disposed of. Table 5 presents the results of leaching tests conducted by Starmet on several common forms of DU, DUAGG, and DUCRETE. The leach test used was the EPA Toxic Characteristic Leaching Procedure (TCLP), which is used to assess heavy metal risks to the environment. Although uranium is not regulated as a toxic heavy metal under the Resource Conservation and Recovery Act (RCRA), it is a toxic metal. The potential effects on groundwater should be considered in the case of any exposure to the environment in a shielding product or in a disposal cell. The data in Table 5 show that the conversion of UO3 into DUAGG reduces the leaching by 1700- fold, from 6900 mg-U/L to 4 mg-U/L. These results also compare DUAGG and DUCRETE to other solid forms of uranium.

Table 5. Comparative leach test results for DU subjected to EPA TCLP testing

Uranium form <sup>a</sup>	U concentration in leachate (mg-U/L)
DUCRETE	0.42
DUAGG	4
$UO_2$	172
$U_3O_8$	420
UF <sub>4</sub>	7367
$UO_3$	6900

<sup>&</sup>lt;sup>a</sup>The DUAGG and U<sub>3</sub>O<sub>8</sub> were manufactured at Starmet CMI from SRS UO<sub>3</sub>. The UF<sub>4</sub> was converted from UF<sub>6</sub> at Starmet. The UO<sub>3</sub> is from the DOE Savannah River Site and was recovered from reprocessing.

Encapsulating DUAGG into cement further reduces the leaching results by another order of magni-

tude to 0.42 mg-U/L., which is still above the 0.03 mg-U/L drinking water limit in 40 CFR 141–142 within the repository horizon.

# 2.8 Potential DUCRETE Applications

The effectiveness of DU in nuclear shielding for spent fuel and HLW is apparent. While uranium metal may be cost-prohibitive for many shielding applications, INEEL cost estimates show DUCRETE shielding to be cost-effective.<sup>10</sup>

# 2.8.1 Reactor Spent Fuel Storage

Sierra Nuclear developed a conceptual model of its VSC-24 storage casks with DUCRETE. This dry-storage cask will store 24 pressurized water reactor (PWR) or 61 boiling water reactor (BWR) fuel assemblies. Through a series of calculations, Sierra Nuclear showed that such a storage cask is about 30% lighter than one made of ordinary concrete and has a much smaller footprint on the storage pad. The DUCRETE storage cask had a diameter of 89 in. (226 cm), compared with 132 in. (335 cm) for standard heavy concrete. The DOE Integrated Data Base Report for 1994<sup>17</sup> estimates the potential cumulative number of SNF

assemblies expected from PWRs and BWRs through the year 2020. The projected numbers of SNFs from both of these light-water reactors (LWRs) are presented in Table 6. These data are presented graphically in Fig. 13.

These data show that in the conservative, low case as many as 268,000 total LWR SNF assemblies will be released. These assemblies require dry storage in some type of interim storage facility. These dry-storage facilities will be at the reactor sites and probably at the

Table 6. Estimated accumulated number of spent nuclear fuel assemblies through 2020 for the low case

	PWR	BWR	
Year	assemblies	assemblies	Total
2002	69,900	89,700	159,600
2003	72,900	93,000	165,900
2004	75,700	96,800	172,500
2005	79,200	100,000	179,200
2006	81,900	104,700	186,600
2007	84,900	108,700	193,600
2008	88,400	111,600	200,000
2009	90,700	116,300	207,000
2010	93,800	120,400	214,200
2011	96,500	123,800	220,300
2012	98,800	129,500	228,300
2013	102,900	132,800	235,700
2014	105,800	139,400	245,200
2015	107,300	140,800	248,100
2016	109,900	144,200	254,100
2017	111,700	145,500	257,200
2018	113,300	147,900	261,200
2019	114,900	149,400	264,300
2020	116,400	151,600	268,000

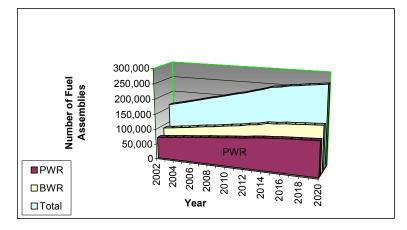


Fig. 13. Estimated cumulative number of SNF assemblies for PWR and BWR light-water reactors through 2020, low-case scenario.

geological disposal site as well. Many current interim dry-storage facilities use concrete casks or silos to hold their SNF. Because of the events of September 11, 2001, there will be a strong push to

accelerate the removal of the current large backlog of LWR SNF assemblies from their current storage in at-the-reactor wet-storage pools.

Studies of the use of DUCRETE in these silos showed that as much as 55.68 tonnes of DUO2 would be used in the DUAGG for each dry-storage cask or silo. This is the equivalent of 49.09 tonnes of DU, or 72.59 tonnes of DUF<sub>6</sub>, for each cask or silo. Given the conservative assumptions of 24 PWR and 61 BWR assemblies per cask or silo, the cumulative amount of DUO<sub>2</sub> that could be used in these dry-

storage casks though 2020 is 408,455 tonnes. This translates into 360,058 tones of DU or 532,473 tonnes of UF<sub>6</sub> to be used in the shielding of SNF assemblies in dry-storage casks or silos. Figure 14 illustrates the potential cumulative use of

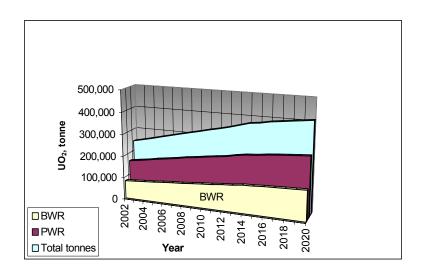


Fig. 14. Potential cumulative use of DUO<sub>2</sub> in SNF storage casks through 2020. Twenty-four PWR and 61 BWR assemblies per storage silo are considered.

408,455 tonnes DUO<sub>2</sub> in these SNF storage casks.

# 2.8.2 Transportable Storage Cask

Current concrete storage systems are too large to transport by rail. With the reduced diameter possible



Fig. 15. GNB CONSTOR cask for storage and transportation of RBMK spent fuel.

with a DU shield, a concept for a transportable cask evolved. The normal unrestricted rail transportation limit in the United States is 128 in. (325 cm). A DUCRETE cask could be used advantageously at the reactor site until an interim spent fuel storage site becomes available—at the DOE's Yucca Mountain facility or elsewhere. From a DUCRETE storage cask used at the reactor's site, the spent fuel could be loaded into a licensed transportation cask and shipped to its destination. Then, the DUCRETE storage cask could be shipped empty to that destination on a rail car and be used to store the spent fuel again. This would save having to duplicate storage capacity at the interim storage site or having lag storage at the DOE disposal site.

Gesellschaft für Nuklear-Behälter (GNB) manufactures a cask, the CONSTOR, which uses reinforced heavy concrete between two 4-cm-thick steel shells. 18 This cask is shown in Fig. 15. The cask has been developed, tested, and licensed according to International Atomic Energy Agency (IAEA) criteria for storage and transportation for the SNF from Russian graphite-moderated boiling water (RBMK) and pressurized water (VVER) reactors.

Using the current heavy concrete casks, barium sulfate requires a concrete-wall thickness of 17.4 cm (with a density of 4.1 gm/cm³). For the same shielding effectiveness (gamma attenuation), DUCRETE concrete with a density of 6.0 g/cm³ needs a wall thickness of only about 11.5 cm. Assuming that the other structural features of the CONSTOR were not compromised by the reduction in wall thickness, this cask might be redesigned to hold more fuel while maintaining the same external diameter. Furthermore, the thinner and denser DUCRETE walls can conduct more heat and reduce the inner wall temperatures of the casks, facilitating the storage and transport of "fresher" SNF bundles.

A capacity increase is a major factor in reducing the overall storage and transport cost to utility companies. Alternatively, if the capacity increase is not needed, the diameter reduction (about 6 cm, or 35%) will reduce the costs of materials and fabrication as well as the overall system weight. The thinner DUCRETE shell would also allow higher heat loadings in these casks. Based on a very simple model, the estimated weight reduction for the CONSTOR cask with DUCRETE would be about 10 tonnes.

# 2.8.3 Disposal Waste Package Shield

Currently, the Yucca Mountain Project (YMP) bans the use of concrete in the repository. This is based on the perception that cement can affect the pH of the groundwater and therefore increase the mobility of nuclides and the corrosion of repository disposal packages. It is our experience that pozzolanic-based cements buffer the pH and greatly decrease the mobility of nuclides, including uranium and long-lived actinides. Furthermore, pozzolanic concretes with excess silica (see Table 4), such as those proposed for DUCRETE, can be formulated to promote the formation of passivating films to protect the metals from corrosion.

Therefore, DUCRETE casks could be used at the repository for shielding the waste package, provided special pozzolanic cements are used (Fig. 16). Such special concretes have been developed for other

# **DUCRETE Overpack System**

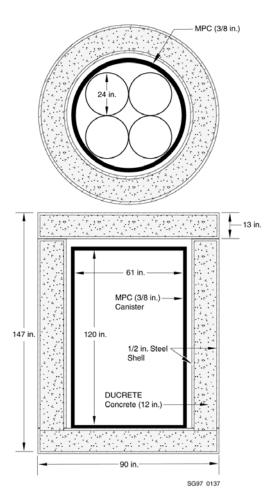


Fig. 16. High-level waste (HLW) canister shield developed for a multipurpose HLW canister.

geological disposal applications. In the YMP concept, the waste disposal package is a key engineered feature needed to ensure that the repository meets its long-term performance criteria. The waste package is the container into which the spent fuel will be loaded at the repository. For repository use, this package must be fabricated from materials that are designed to last 10,000 years. Over the past six years, design specifications for waste packages have increased the external dose specification from about 100 R/h to over 1000 R/h.

A DUCRETE cask could be fabricated that would integrate the waste package and required shielding capabilities needed at the repository. Calculations by J. Tang showed that DUCRETE shielding factors are 20 times better than those of other materials evaluated.<sup>19</sup> Such a multiple-purpose cask would allow the repository to be designed for unrestricted human access for fuel emplacement, inspections, and underground facility maintenance. A paper by Quapp describes this concept, its benefits, and its costs.<sup>20</sup> Overall, the use of DUCRETE shielding for utility storage, as well as for shielding waste packages, would simplify operation of the repository and provide an answer to the long-term disposition of the huge inventory of DU.

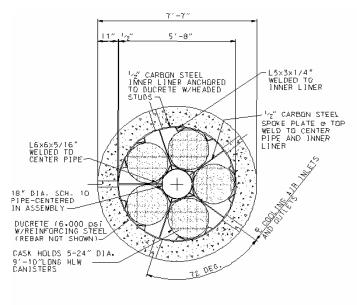
There are a number of benefits from using a pozzolanic-based DUCRETE in the repository:

- The material provides a physical barrier to rock falls and drift convergence.
- Pozzolanic additives passivate steel and nickel surfaces and reduce corrosion rates.
- DUCRETE provides high chloride and sulfate resistance with impermeable hydroaluminosilicate phases.
- A pozzolanic-based DUCRETE buffers geochemistry to a mild alkaline pH range.
- A mild steel fiber and rebar reinforcement buffers Eh to a reducing environment that stabilizes insoluble UO<sub>x</sub>s.
- DUO<sub>2</sub> and pozzolans reduce the solubility and transport of uranium, fission products, and transuranics.

# 2.8.4 High-Level Waste Canister Shielding

The current U.S. practice for shielding vitrified HLW is to store canisters in a shielded storage building. These canister are 24 in. (61 cm) in diameter and about 10 ft (305 cm) long. The cost of the storage building for these canisters in 1990 was about \$100 million. A variety of studies<sup>21</sup> have shown that DUCRETE casks may provide a more cost-effective solution for future storage requirements. The existing glass waste storage buildings have sufficient capacity for about 40% of the projected production of canisters. A capacity equivalent to 1.5 times the present building size is needed for the projected canister production.

As part of a proposal to the Savannah River Site, Starmet and Duke Engineering and Services evaluated the feasibility of using DUCRETE concrete material for HLW storage casks.<sup>22</sup> This cask concept is shown in Fig. 17. Use of DUCRETE allows the wall thickness to be kept to about 10 in. and results in an external radiation field of 10 milliroentgen/h at a 2-m distance from five HLW canisters. The cask uses a stainless steel glass canister as containment and natural convection cooling to remove the thermal energy. This cask is intended to be a storage-only cask for use at the Savannah River Site as an alterna-



tive to a second glass waste storage building. Cost estimates indicate that for an equivalent storage facility, including pad and handling facilities, the DUCRETE cask storage system would be less expensive than a new glass waste storage building. In addition, the casks would use most of the 55 million pounds of unneeded  $UO_x$  on the Savannah River Site, thus saving more than \$80M in disposal costs.

At the end of these casks' storage lives, they could be transported separately from the HLW and used for shielding at the repository, when use of concretes is allowed in the final repository. Since DUCRETE is an excellent waste form, another alternative would involve disposing of the silos into shallow land burial and even using them there as waste burial vaults.

# 2.8.5 Remotely Handled Transuranic (TRU) Waste Drum Overpack

In 1997, INEEL identified a need for a large number of shielded drum overpacks. Starmet designed and built DUCRETE drum overpacks as part of its internally funded product development (Fig. 18). These overpacks use a stainless steel shell (14-gage) with DUCRETE poured between the walls and within the lid.

# 2.9 Current DUAGG Testing

Current testing at ORNL measures the extent and rates of surface reactions of DUAGG under the expected service temperatures and the simulated chemical environments of cement pastes. The DUO<sub>2</sub>/basalt aggregates in DUAGG samples from Starmet are cleaned in an ultrasonic bath with distilled water to remove surface residuals. These DUAGG samples are then tested for aggregate reactivity using the American Society for Testing and Materials (ASTM) C289-94 method<sup>23</sup> to measure interactions with pore liquids that are anticipated in concrete pastes. The almond-shaped DUAGG aggregates are 1.59 cm long, 0.95 cm wide, and 0.64 cm thick. The surface area is estimated by assuming a prolate spheroid shape of 3.5 cm<sup>2</sup> or 2.8 cm<sup>2</sup> using a parallelepiped model. Measurements on 17 DUAGG pellets arrived at an average surface area of  $5.77 \pm 0.01$  cm<sup>2</sup>.



Fig. 18. Stainless steel-clad DUCRETE drum overpack.

At a consistent surface-to-liquid ratio of 1:10, the sintered DUAGG samples are exposed to distilled water, to a 1N sodium hydroxide standard solution, and to an extract of high-alkali cement water. The exposure temperatures are 25, 66, and 150°C for six time intervals of 30, 60, 90, 180, 240, and 360 days. At the end of each exposure period, the vessels in which the samples are immersed are opened and the liquid phase is separated and filtered for analysis of silica, uranium, alkalies, and other elements measurable by ion coupled plasma (ICP) mass spectrometry. The elemental quantities leached are compared to determine the degree of erosion of the aggregate surfaces.

The surfaces of the exposed aggregates are compared, according to the guidelines of ASTM C295-98, <sup>24</sup> using SEM and energy-dispersive X-ray fluorescence (EDX) analyses. The extent of surface

alteration and formation of secondary minerals are used to indicate the potential for aggregate paste interactions in DUCRETE.

For 25 pellets with an estimated density of approximately  $8.5 \text{ g/cm}^3$ , the average weight was calculated to be  $6.39 \pm 0.01 \text{ g}$ , while the average volume was found to be  $0.784 \pm 0.002 \text{ cm}^3$ . The measured density of a DUAGG pellet is  $8.15 \text{ g/cm}^3$  and the surface area is  $5.8 \text{ cm}^2$ . For a leachate volume/ surface ratio of 10, the volume of liquid used in the vessels for two aggregate pellets is approximately 58 mL. The vessels are rated for 250 psig, and the maximum pressure for a vessel kept at  $150 ^{\circ}\text{C}$  is less than 160 psig. The vessels are enveloped in aluminum foil in a secondary container prior to being placed in the oven to prevent leakage in case of a failed rupture disk or vessel gasket. These measurements at ORNL began in September 2001. The ICP analyses for the 30- and 60-day tests are completed, and the microscopic examinations are under way.

# 2.10 DUAGG and DUCRETE Development Needs

A detailed development proposal was developed in 1998<sup>25</sup> and updated in 2001. Highlights from that plan are as follows:

- Develop testing protocols for comparing the interactions between DU aggregate and the cement paste in the heavy concrete.
- Acquire DUAGG samples needed for tests.
- Provide a basis for economic evaluation and component design optimization by measurement of key physical properties of DUAGG matrices, such as thermal conductivity and expansion, shielding parameters, density, and strength.
- Initiate a long-term test program to evaluate the stability of DU aggregate in various matrices under elevated and cyclic temperatures between 20 and ~1400°F. This demonstration of the stability of the DU aggregate within an oxidizing matrix is needed to provide confidence that DUAGG casks will be functional for the decades of their required service lives. Limited-duration tests have shown instability within 30 days, but at temperatures above the expected operating temperature of the DUAGG matrices. These activities are critical because a long period of time is required for each exposure test.
- Evaluate the technology for making large DUAGG matrix shapes that have adequate homogeneous aggregate distribution and compressive strength and density. This evaluation is needed to demonstrate that large, thick DUAGG matrix shapes have a consistent and uniform mixture of binder and aggregate throughout the shielding and that they can be manufactured to the required performance specifications.
- Establish the use of microfiber reinforcement to increase energy-absorption capacity to better survive assault scenarios.
- Assess alternative disposal options for heavy concrete. This will ensure that heavy DUAGG shapes have a feasible and cost-effective disposal path at the end of their useful lives as shields.
- In collaboration with a potential vendor, construct and test a full-scale DUAGG matrix storage cask.
- Assess the regulatory and econometric market for DUCRETE storage casks.

# 3. DUPoly

Brookhaven National Laboratory (BNL) has completed preliminary work to investigate the feasibility of encapsulating DU in low-density polyethylene to form a stable, dense product known as DUPoly. <sup>26, 27</sup> DU loadings as high as 90 wt % were achieved using DU aggregate in a technique known as micro- or macro-encapsulation (Fig. 19).

BNL has used polyethylene encapsulation through the coextrusion of a mixture of polyethylene and waste materials such as salts and contaminated soils for many years. Encapsulating DU oxide or UF<sub>4</sub> in the polyethylene matrix was an extension of their experience.

Therefore, the treatment of DU materials by polyethylene encapsulation is technology with a proven record in processing powder and granular materials. In addition, the process is very flexible. Polyethyl-

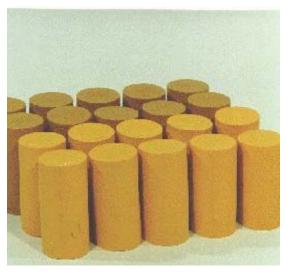


Fig. 19. DUPoly samples produced at BNL containing 80 wt % depleted UO<sub>3</sub>.

ene products can be heated and reworked if future needs change. DU can also be retrieved from DUPoly by thermal processing if needed as a resource in the future. Recycled plastics from industrial or post-consumer sources can be used in place of virgin materials to reduce costs and produce valuable products entirely from recycled materials. BNL has extensively developed, tested, and demonstrated the polyethylene encapsulation processes for low-level radioactive, hazardous, and mixed wastes. During processing, waste materials are mechanically mixed into the molten polyethylene binder, producing a workable homogeneous product. The process is not susceptible to chemical interactions between the waste and binder, enabling a wide range of acceptable waste types, high waste loadings, and technically simple processing under heterogeneous waste conditions. The process has evolved from proof of principle, through bench-scale development and testing, to full-scale technology demonstration.

A prototype DUPoly transportation and disposal cask has recently been fabricated to facilitate the disposal of a highly radioactive radium-beryllium source being stored at BNL. DUPoly is ideally suited for reducing both gamma and neutron radiation from the source. The prototype DUPoly transportation and disposal cask will allow a one-time transfer of the source into the shielded cask, shipment to the Hanford disposal site in compliance with U.S. Department of Transportation standards, and burial—all without having to remove the source or handle it again. The DUPoly cask is constructed from recycled materials, so it is inexpensive and can be economically used for disposal.

BNL recently published a patent that gives performance data on the material. A plot of density versus uranium loading is shown in Fig. 20. The maximum product density of  $4.2 \text{ g/cm}^3$  was achieved using  $UO_3$ , but use of  $UO_2$  would increase product density to an estimated  $6.1 \text{ g/cm}^3$ . Use of DU aggregate through micro- or macro-encapsulation would provide additional product-density improvements of up to about  $7.2 \text{ g/cm}^3$ .

BNL also conducted compression tests of the DUPoly using  $2 \times 4$  in. cylinders. The compression test data from the patent is shown in Fig. 21.

The melting point of polyethylene is 140°C, and its maximum long-term service temperature is 100°C. If used, DUPoly must be encased in a design that mechanically meets the service criteria without a physical contribution from the shield materials. Also, UO<sub>3</sub> is a mild oxidizer that can release oxygen when heated in a reducing environment. Therefore, fine particles and high temperatures must be avoided.

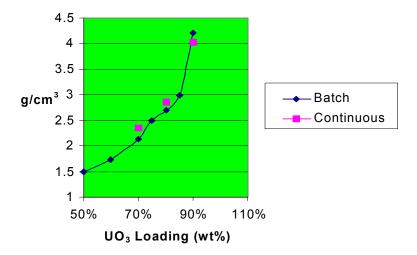


Fig. 20. DUPoly density (in g/cm<sup>3</sup>) as a function of UO<sub>3</sub> loading. "Batch" and "continuous" refer to two types of UO<sub>3</sub> production processes. The processing rate was about 16 lb/h.

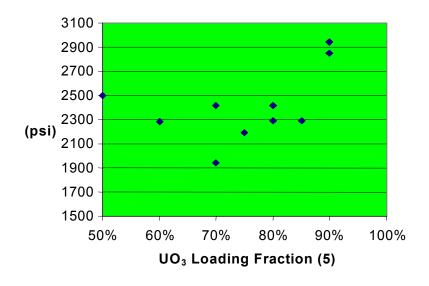


Fig. 21. DUPoly compressive strength (psi) as a function of UO<sub>3</sub> loading.

# 4. PYRUC

Pyrolytic uranium compound (PYRUC) is a shielding material consisting of DU (UO<sub>2</sub> or UC) in either pellet or microsphere form, coated or uncoated, and combined with a solid matrix binding material. Microspherical products have been formed by treating uranyl nitrate by gelation methods. <sup>32, 33, 34</sup> A number of binding materials are possible (e.g., tar or pitch precursor and high-temperature resins). Murray and his colleagues also considered cement as a binder material.

The production of coated  $\mathrm{UO}_x$  microspheres is an established technique that was developed in the mid- to late 1960s as a potential alternative method for fabricating nuclear fuel for reactors. Figure 22

illustrates the production and shielding concepts. Thus, mixing microspheres with binders will clearly produce a matrix with both high-Z and low-Z materials for shielding against both gamma and neutron radiation.

This initial proposal suggested using the sol-gel process for producing the carbon particles. There are much less costly alternative routes to producing carbon particles. However, because of limited funding, only conceptual studies of PYRUC have been performed. No materials have been fabricated, and thus, no performance data are available. For systems using tar or pitch, the density would be expected to be comparable to that of DUPoly.

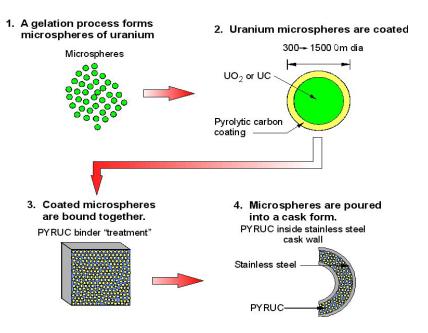


Fig. 22. PYRUC process flow diagram.

# 5. FUTURE RESEARCH AND DEVELOPMENT NEEDS

Future development efforts should be selected on the basis of the potential uses in storage, transport, and disposal casks. Both DUCRETE and DUPoly have undergone sufficient development to allow reasonable estimates of performance attributes for spent fuel and high-level waste shielding applications. Further radiation attenuation measurements should be made using the gamma and neutron spectrums expected from SNF and HLW. Additional analytical studies should be conducted to determine the optimum shielding performance for various loadings of DU in particular binders. The effectiveness of adding additional neutron attenuation elements to the aggregate-binder mixture needs to be evaluated, particularly in the case of materials with fast neutrons.

Demonstrations of the large-scale manufacture of aggregates and tests of prototype casks are needed to establish the technical feasibility of using these materials and to convince cask manufacturers that these new materials are worth their efforts. Finally, the efficacy and benefits of these technologies need to be demonstrated to the potential customers and users of these casks.

# 6. CONCLUSIONS

Shielding materials containing DU provide beneficial uses for a large portion of the DU inventory. Two of these materials, DUCRETE and DUPoly, have been successfully tested with small quantities. The third, PYRUC, is a hybrid of both that uses the sol-gel production process for the DU aggregate.

Both DUCRETE and DUPoly provide a shield matrix with high-Z and low-Z shielding attributes when optimized. Their properties allow a reduction in the size and weight of storage, transport, and disposal casks. However, further development will be required if structural strength is needed in a cask design.

Research on these materials is needed to complete the optimization of gamma and neutron absorbers and to qualify them for licensed nuclear applications. Lower-cost production processes also need to be developed. It is still possible to qualify these materials for use in final geological disposal at YMP because the final design and licensing of the YMP repository is incomplete. While experimental work needs to be completed to demonstrate that these materials can meet the disposal site's shielding requirements, there may very well be adequate time to prove that these technologies are mature enough to be applied at YMP.

DU materials can provide a large performance advantage over existing shielding materials. Commercial cask firms interested in proposing technologies have not done so because the fabrication technologies are not yet mature. Their biggest concerns are related to manufacturing costs and regulatory acceptance.

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