

# Characterization of nanoparticles by continuous contrast variation in SAXS

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# Chapter 1

## Introduction: Nanoparticles in medicine and biology

### 1.1 Polymeric colloids

#### 1.1.1 Functionalization for protein binding

#### 1.1.2 Polymerization consequences

initiator, co-monomer, surfactants

### 1.2 Liposomal nanocarriers

formation from amphiphilic lipids

#### 1.2.1 Phospholipid bilayer

typical lipid HSPC, DPPC, cholesterol, PEG

#### 1.2.2 polydispersity control

extrusion, paper with zoltan about scattering in SSLs

#### 1.2.3 Drug carrier and SSLs

stealth function, bilayer stability, filling with pH gradient

## **1.3 Physicochemical characterization**

### **1.3.1 Dimensional metrology and traceability**

### **1.3.2 Characterization tools**

#### **Single-particle method**

AFM, TEM, SEM, TSEM

#### **Ensemble methods**

DLS, DCS, SAXS

# Chapter 2

## Theoretical Background

### 2.1 Interaction of light and matter

#### 2.1.1 X-ray cross sections

#### 2.1.2 Rayleigh and Mie scattering

### 2.2 Small-angle X-ray scattering

#### 2.2.1 Physical process

#### 2.2.2 Evaluation of the scattering intensity

Form factor \* S(q) Electron density Number of colloids

What is  $q$ ?

Modelling of the scattering curve

What about size distributions? Log-normal, gaussian, Monte-carlo free number of sizes (Pauw)

The scattering intensity of an ensemble of randomly oriented nanoparticles in suspension can be expressed as a function of the momentum transfer  $q$ , modulus of the scattering vector  $\vec{q}$ , as

$$I(q) = N \int_0^\infty g(R) |F(q, R)|^2 dR, \quad (2.1)$$

where  $N$  is the number of scatterers,  $g(R)$  is the size distribution function and  $F(R)$  is the particle form factor, which depends on the inner radial structure of the particle. If the particle shows a heterogeneous morphology, the form factor differs qualitatively for different suspending medium densities. For sufficiently monodisperse particle suspensions, the Fourier region of the scattering curve shows pronounced minima that characterize the particle structure.

For a typical morphology with sharp interfaces between the radial symmetric components of the particle with radius  $R_i$  the form factor is

$$F(q, R) = \Delta\eta f_{sph}(q, R) + \sum_{i=1}^{n-1} \Delta\rho_i (f_{sph}(q, R_{i+1}) - f_{sph}(q, R_i)), \quad (2.2)$$

where  $R$  is the external radius of the particle,  $n$  is the number of concentric shells and  $f_{sph}$  is the form factor of a homogeneous solid sphere given by

$$f_{sph}(q, R) = \frac{4}{3}\pi R^3 \left( 3 \frac{\sin qR - qR \cos qR}{(qR)^3} \right). \quad (2.3)$$

Sphere Gudrun polymeric colloids???

In the case of the PMMA-COOH colloids, the form factor is calculated for a homogeneous solid sphere with electron density  $\rho_0$ :

$$f_{sph}(q, R) = \frac{4}{3}\pi R^3 \left( 3 \frac{\sin qR - qR \cos qR}{(qR)^3} \right) = \frac{F(q, R)}{\rho_0}. \quad (2.4)$$

Core-shell Interface effects???

The model represents a radially symmetric particle with a sharp interface between the outer shell and the inner core. The form factor is described by

$$F(q, R) = \Delta\eta f_{sph}(q, R) + \Delta\rho [f_{sph}(q, R) - f_{sph}(q, R_{core})], \quad (2.5)$$

where  $R$  and  $R_{core}$  are the outer shell and inner core radii respectively, the excess of electron density is  $\Delta\rho = \rho_{shell} - \rho_{core}$  and the contrast is expressed as  $\Delta\eta = \rho_{core} - \rho_{solv}$ , where  $\rho_{solv}$  is the electron density of the suspending medium.

Onion model It can be used for single-SAXS experiment maybe

Vesicle 5 gaussian????

inclusion of background  $a+b*q^{-4}$



## Guinier approximation

deviation when using too few point Polydispersity effects

## 2.3 Contrast variation

Solvent variation

ASAXS

When analyzing contrast variation data, a widespread theoretical approach is based in the non-interacting model proposed by Stuhrmann & Kirste 1965; 1967 for monodisperse particles. The so-called *basic functions* formulation differentiates, independently of the particle inner structure, the contributions which depend on the varying solvent density or contrast ( $\Delta\eta = \rho_{core} - \rho_{solv}$ ) and on the excess of electron density of each component  $\Delta\rho_i = \rho_i - \rho_{core}$ .

### 2.3.1 Isoscattering point

One of the best known features appearing in a contrast variation experiment is the existence of *isoscattering points*. At these specific  $q$ -values, the scattering intensity is independent of the adjusted solvent contrast, i.e. all scattering curves intersect in the isoscattering points regardless of the contrast. The isoscattering points  $q^*$  are particularly interesting because they emerge for any spherical particle with an inner structure and a sufficiently narrow size distribution. From the contrast-depending part of equation (2.2), a model-free expression can be derived which relates the position of the isoscattering points  $q_i^*$  with the external radius of the particle  $R$ , independent of its radial structure Kawaguchi & Hamanaka (1992):

$$\tan(q_i^* R) = q_i^* R \quad (2.6)$$

The positions of the isoscattering points correspond to the minima positions of the scattering intensity of a compact spherical particle with radius  $R$ . Although this expression is derived for the monodisperse case, it can still be applied up to a moderate degree of polydispersity, if care is taken regarding the shift of the minima position due to polydispersity Beurten & Vrij (1981). If defining the polydispersity degree  $p_d$  as the full width half maximum of the particle size distribution divided by its average value, for size distributions with  $p_d$  larger than  $\approx 30\%$ , the isoscattering point is not well defined and the intersection point of the curves is smeared out, showing a diffuseness in the isoscattering point position Kawaguchi & Hamanaka (1992).

## Possible deviations

Polydispersity and ellipticity smearing (simulation, calculation)

### 2.3.2 Basic functions approach

When analyzing contrast variation data, a widespread theoretical approach is based in the non-interacting model proposed by Stuhrmann & Kirste (1965; 1967) for monodisperse particles. The so-called *basic functions* formulation differentiates, independently of the particle inner structure, the contributions which depend on the varying solvent density or contrast ( $\Delta\eta$ ) and on the excess of electron density of each component of the particle.

Deriving from this approach, the scattering intensity can be expressed as the combination of contributions corresponding to different features of the particles:

$$I(q) = I_c(q) + \Delta\eta I_{sc}(q) + (\Delta\eta)^2 I_s(q) \quad (2.7)$$

The  $I_c$  function contains the contributions from the density fluctuations inside the particle, the contribution  $I_s$  is the so-called *resonant term* and  $I_{sc}$  is the cross-term function.

## Shape factor

The  $I_s(q)$  function, also known as *shape factor*, corresponds to the scattering contributions from particles with homogenous density and a size equivalent to the volume inaccessible to the solvent. By modelling the shape factor function, the shape and size distribution of the polymeric colloids can be determined independently of their inner structure.

For this purpose, a spherical form factor for homogeneous colloids with a gaussian size distribution was utilized, similarly to the PMMA-COOH example. In order to obtain the particle sphericity, an ellipsoid model was employed.

## Guinier law

Gyration radius

The radius of gyration  $R_g$  is systematically employed in small-angle scattering as an evaluation tool Mertens & Svergun (2010); Sim *et al.* (2012). It can be calculated using the Guinier approximation Guinier (1939); Guinier & Fournet (1955), which

assumes that the scattering intensity behaves in the limit of small  $q$  as

$$I(q) = I(0) \exp \left( -\frac{R_g^2}{3} q^2 \right), \quad (2.8)$$

where  $I(0)$  is known as forward scattering or intensity at zero angle. Using the basic functions approach, the radius of gyration of a monodisperse, heterogeneous particle can be expressed as a function of the solvent electron density  $\rho_{solv}$  and the average electron density of the particle  $\rho_0$  Feigin & Svergun (1987)

$$R_g^2 = R_{g,c}^2 + \frac{\alpha}{\rho_0 - \rho_{solv}} - \frac{\beta}{(\rho_0 - \rho_{solv})^2}, \quad (2.9)$$

where  $R_{g,c}$  is the radius of gyration of the particle shape corresponding to the volume inaccessible for the solvent  $V_c$ ,  $\alpha$  characterizes the distribution of different phases inside the particle and  $\beta > 0$  considers the eccentricity of the different scattering contributions Stuhrmann (2008). Nevertheless, particle aggregation influences the scattering curves especially in the Guinier region and must be explicitly avoided.

Avdeev (2007a) proposed an extended version to equation (2.9) for the case of a polydisperse particle ensemble by introducing the *effective* values  $\tilde{R}_{g,c}^2$ ,  $\tilde{\alpha}$  and  $\tilde{\beta}$ , which are the intensity-weighted averages of the corresponding parameters over the polydispersity. The observed average electron density is not affected by the polydispersity ( $\tilde{\rho}_0 = \rho_0$ ) if the volume ratio between the different particle components is constant for all particles in the ensemble.

Assuming the same premise, the intensity at zero angle is given by

$$I(0) \propto N (\rho_0 - \rho_{solv})^2, \quad (2.10)$$

with a minimum at  $\rho_{solv} = \rho_0$ . Therefore, by analyzing the Guinier region of the scattering curves, the average electron density of the particle can be obtained without assuming an *a priori* inner structure.

Using the models presented above, it is possible to obtain by independent means the external radius and the average electron density of the particle in suspension.

$I(0)$

what happens in polydisperse systems?

# Chapter 3

## Experimental setup for SAXS measurements

### 3.1 BESSY II

### 3.2 FCM Beamline

#### 3.2.1 Transmission measurements

calibrated diodes, SYRES II????

### 3.3 Small-angle X-ray scattering

The measurements were performed at the four-crystal monochromator beamline in the PTB laboratory at the electron storage ring BESSY II (Berlin, *Germany*), which provides highly intense, collimated synchrotron radiation focused on the sample and collimated into a 0.5 mm circular spot by Ge pinholes situated between the sample and the monochromator with an energy resolving power  $E/\Delta E$  of  $10^4$ . To measure the total flux and sample transmission, photodiodes were used which were calibrated against a cryogenic electric substitution radiometer with a relative uncertainty of 1 % Krumrey & Ulm (2001).

The rectangular capillary is placed in a sample holder which allows the movement with micrometer precision in the directions perpendicular to the incoming beam, as depicted in figure ?? . In order to determine the central vertical capillary axis, a horizontal X-ray transmission scan is performed at two different vertical positions

of the capillary spaced by 20 mm. The central vertical axis can be drawn from the centers of both measurements and the sample can be moved along this axis by the simultaneous operation of the vertical and horizontal motors.

The sample was moved in steps of 0.5 mm along the central vertical capillary axis and exposed at each position for 45 seconds. At these positions, the solution transmittances were previously measured and the suspending medium electron density calibrated. The measured scattering curve is an average over a range of solvent electron densities associated with the beam size. The momentum transfer  $q$  of the scattering curves was calculated using

$$q = \frac{4\pi E}{hc} \sin \theta, \quad (3.1)$$

where  $\theta$  is half of the scattering angle,  $h$  is the Planck constant and  $c$  is the speed of light. The incident photon energy  $E = (8800.0 \pm 0.8)$  eV was chosen to be higher than the photon energy for the transmission measurements to improve the recorded statistics, due to a ca. 150 higher transmission Henke *et al.* (1993). The scattered X-ray photons were collected with a vacuum-compatible Pilatus 1M hybrid-pixel detector (Dectris Ltd, (Baden, *Switzerland*)) with a pixel size of  $d = (172.1 \pm 0.2)$   $\mu\text{m}$  at a distance  $L = (4540.2 \pm 0.8)$  mm from the capillaries, determined by triangulation using a calibrated length measurement system Wernecke *et al.* (2014). The obtained scattering curve was normalized to the exposure time and the incident intensity, measured by means of a calibrated transparent silicon diode. In total, 40 scattering curves with different solvent electron densities were measured at two different times  $t_1 = 78$  min and  $t_2 = 156$  min after filling the capillaries.

### 3.3.1 Pilatus detector

high dynamic range noise free

### 3.3.2 HZB SAXS setup

distance calibration

10-4 uncertainty

### **3.3.3 Radial integration and error propagation**

### **3.3.4 Absolute intensity calibration**

#### **Flux monitor**

thin diode

#### **Detector efficiency**

pilatus and thin diode

## **3.4 Continuous contrast variation**

### **3.4.1 Filling of capillaries**

galden at bottom, reference layer

#### **Capillary homogeneity**

Hilgenberg

### **3.4.2 Calibration of solvent density and finding of main axis**

The transmitted intensity through the sample is recorded at a photon energy of  $E = (5500.0 \pm 0.5)$  eV for 10 seconds at each position. The measurement consists of 20 points spaced 0.5 mm along the central vertical axis of the capillary. The overall X-ray transmission measurement requires approximately 5 minutes, which is within the calculated diffusion timescale of the aqueous sucrose solution. The solvent electron density profile within the density gradient capillary derived from this measurement is depicted in figure ???. A uniform thickness of the capillary within 0.5 % along this axis was determined by measuring the X-ray transmission of an empty capillary. The associated uncertainty in the sample transmission measurement is below 4 %. The sample thickness is assumed to be constant. This transmission measurement is performed both immediately before and after recording the scattering patterns, which takes 15 minutes to complete. The transmittance values used for the density calibration are then linearly interpolated between both data sets taking into account the time-dependence. These values can be converted to solvent electron densities via

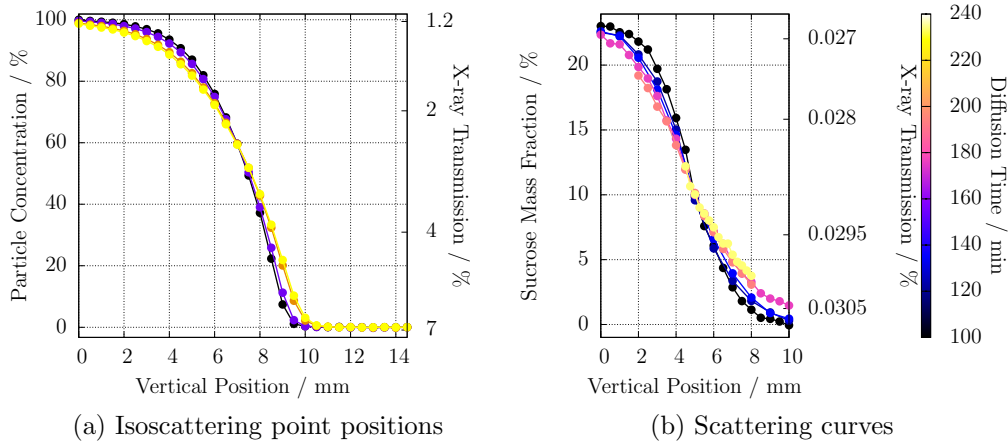


Figure 3.1: Typical measurement of particles with different diffusion timescale: Sucrose (with Kisker NPs) measured at 5500 eV and Colloids (Ludox HS40) measured at 8000eV

the Beer-Lambert law, which relates the density of the solution with the transmitted intensity:

$$\rho(z) = A (\ln I_0 - \ln I(z)). \quad (3.2)$$

Here  $\rho$  is the electron density of the suspending medium,  $I$  and  $I_0$  are the transmitted and incoming intensities respectively and  $A$  is a factor determined by the reference values of the solvent electron density at the vertical limits of the capillary at the initial time. The sucrose concentration in solution expressed as the mass fraction  $M$  at these reference points can be converted to electron densities with the empirical formula  $\rho = 1.2681M + 333.19 \text{ nm}^{-3}$  Haynes (2012). The suspending medium electron density shows a maximum uncertainty of  $1 \text{ nm}^{-3}$  associated with the vertical size of the focused X-ray beam.

### 3.4.3 Limitations

#### Density range

sucrose, fructose, iodixanol

#### Challenges with different contrast agents

Background subtraction, induced aggregation by heavy salts

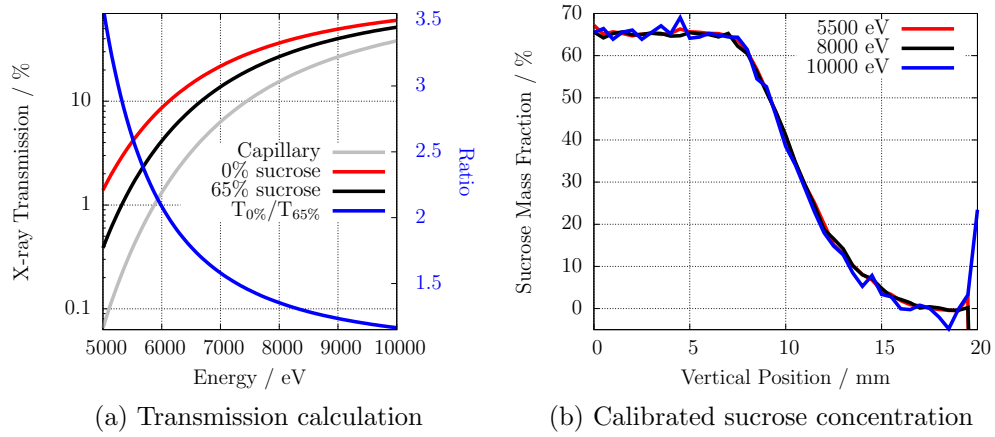


Figure 3.2: Statistics in the transmission measurement depending on the incoming energy. For lower energies, the transmission differences are larger and hence the statistics better. This was measured using only aqueous sucrose 65% in plain water (no colloids on it) Nov 2014

### Comparison to other contrast variation scattering techniques

SANS (deuterated water) RSoXS in polymeric colloids (H.Abe 2006), Carbon K-edge



## Chapter 4

# Contrast variation in SAXS with the density gradient technique

The morphology of nanoparticles determines the properties necessary for their utilization in real-world applications. For instance, in drug delivery devices the phenomena involved in biocompatibility reactions (e.g. protein adsorption) depend on the amount of available surface and the nanoparticles' properties Vittaz *et al.* (1996*a*). Particularly, polymer lattices and biodegradable nanoparticles have been of growing importance of late as drug carriers Kattan *et al.* (1992) and thus extensively characterized Soppimath *et al.* (2001). The size determination and the characterization of the radial structure of the particles are therefore fundamental tasks.

The contrast variation method in Small Angle X-ray Scattering (SAXS) experiments consists in systematically varying the electron density of the dispersing media to study the different contributions to the scattering intensity in greater detail as compared to measurements at a single contrast. It emerges as an ideally suited technique to elucidate the structure of particles with a complicated inner composition and has been repeatedly employed to investigate the radial structure of latex particles suspended in an aqueous medium Dingenouts *et al.* (1999*a*); Ballauff (2011*a*). In Small Angle Neutron Scattering (SANS) the contrast variation technique is widely used by mixing water and deuterium oxide, but the use of deuterated chemicals and the incoherent contribution to the background as well as the limited access to neutrons restrict the application of this technique. Other methods for structural investigation (e.g. transmission electron microscopy Joensson *et al.* (1991*a*); Silverstein *et al.* (1989*a*)) require prior treatment of the sample and are not ensemble averaged.

In SAXS, the solvent contrast variation technique is achieved by adding a suitable

contrast agent to the suspending medium (e.g. sucrose) and recording the scattering data as a function of the adjusted solvent electron density  $\rho_{solv}$  Ballauff (2001); Bolze *et al.* (2003). In order to resolve small changes of the radial structure, the average electron density of the colloidal particles must be close to the dispersant's, i.e., the *match point* should be approached, where the average contrast of the particle vanishes. In the case of polymeric lattices with electron densities ranging from 335 to 390 nm<sup>-3</sup>, an aqueous sucrose solution is very well suited as the suspension medium, due to the easy realization of concentrated solutions with electron densities of up to 400 nm<sup>-3</sup>. Previous studies on globular solutes Kawaguchi & Hamanaka (1992) and the influence of the sucrose on the size distribution of vesicles Kiselev *et al.* (2001) show the feasibility of this technique, while further studies have investigated the effect of the penetration of the solvent into the particles Kawaguchi (1993).

The preparation of a number of different sucrose solutions has been a major inconvenience in solvent contrast variation experiments, due to the tedious, time-consuming process, possible inaccuracy in the sucrose concentration and the discrete range of available solvent electron densities. In this article we propose a novel approach using a density gradient column, which allows the tuning of the solvent contrast within the provided density range, resulting in a virtually continuous solvent contrast variation. By filling the bottom part of the capillary with a particle dispersion in a concentrated sucrose solution and the top part with an aqueous solution of the same particle concentration, a solvent density gradient is initiated with a constant concentration of nanoparticles along the capillary. Density gradient columns are extensively used in fields like marine biology Coombs (1981) or biochemistry together with centrifugation Hinton & Dobrota (1978), to create a continuously graded aqueous sucrose solution by diffusion of the sucrose molecules. Combining this approach with SAXS, it is possible to choose *in situ* the most appropriate solvent densities to perform measurements close to the contrast match point and to acquire extensive datasets in a short interval of time through the high brilliance and collimation of current synchrotron radiation sources. These datasets can be analysed using different, complementary evaluation methods. In this article, both a model-free theoretical framework as well as model fit are applied and, in combination, deliver a detailed insight into the inner structure of particles.

In order to demonstrate the proposed technique, latex nanoparticles with a core-shell structure were measured. The particles have a narrow size distribution and consist of a spherical polystyrene (PS) core enclosed by a thin shell of a denser polymer, most likely poly(methyl methacrylate) (PMMA). This is presented according to the following structure.

Firstly in §, the underlying theory of SAXS contrast variation is briefly reviewed

and the scattering form factor used for the model fitting is presented. The details of the experimental data acquisition are shown in §, followed by SAXS data evaluation using different methods in §4.5, jointly with a discussion of the experimental measurements and a consistency check of the obtained results. Finally, in § the experimental results of the particle size distribution and radial structure are summarized and the applicability of the solvent contrast variation technique in SAXS is discussed.

## 4.1 Materials and Methods

The preparation of a polymeric nanoparticle suspension density gradient within a glass capillary using an aqueous sucrose solution, the X-ray transmittance measurements at different positions along its vertical axis and the collection of scattering patterns at the calibrated capillary positions with distinct contrasts are described in the following sections.

### 4.1.1 Particles and chemicals

Carboxylated polystyrene nanoparticles with a nominal size of 105 nm suspended in water were purchased from Kisker Biotech (Steinfurt, *Germany*). The synthesis by multi-addition emulsion polymerization suggests that the assumption made in §?? is correct and the average density of the particle is not altered by the size polydispersity.

### 4.1.2 Diffusion time and calibration height

The solvent density gradient was prepared in vacuum-proof borosilicate glass capillaries from Hilgenberg (Malsfeld, *Germany*) with a rectangular cross section of  $(4.2 \pm 0.2) \times (1.25 \pm 0.05)$  mm<sup>2</sup>, a length of  $(80 \pm 0.5)$  mm and a wall thickness of ca. 120  $\mu$ m. The bottom end of the capillary was closed by welding and the lower section, up to a height of ca. 1 cm, was filled with Galden®PFPE SV90 from Solvay Plastics (Brussels, *Belgium*). This fluid has an exceptionally high density of 1.69 g/cm<sup>3</sup>, low viscosity and is immiscible with aqueous solutions. Consequently, a uniform interface with the particle suspension is formed at the bottom.

Directly above the Galden fluid, the denser of two mixtures with different solvent densities and an equal particle concentration of 12.6 mg/ml was filled into the capillary using a syringe up to a height of 9 mm. The dense aqueous solution was prepared with 21.23 % sucrose mass fraction (Sigma-Aldrich (Missouri, *USA*)) with a physical density of  $\rho_1 = 1.088$  g/cm<sup>3</sup>, whereas a lighter one was produced without sucrose ( $\rho_2 = 0.997$  g/cm<sup>3</sup>). The light mixture was then filled on top of the aqueous

sucrose solution along ca. 8 mm. By the time the two components come into contact, the density gradient is started with density values  $\rho_1$  and  $\rho_2$ , a total gradient length  $L = 17$  mm and the interface position at  $z_0 = 9$  mm. The calculated diffusion timescale of the solvent density gradient is ca. 10 minutes, considering the diffusion coefficient  $D = 5.2 \cdot 10^{-10} \frac{\text{m}^2}{\text{s}}$  Uedaira & Uedaira (1985); Ribeiro *et al.* (2006) and assuming that convection effects are negligible due to the small length-scale of the capillary Berberan-Santos *et al.* (1997). The time needed for the transfer of the sample into the high vacuum chamber amounts to ca. 1 hour. Within this time duration, the deviation of the solvent density at both ends of the gradient from the initial value can be estimated with an uncertainty below 0.5 %. If the same capillary is measured at different points in time during the diffusion process of the sucrose, several data sets with different solvent densities can be recorded and a very dense data set with a virtually continuous variation in the suspending medium density can be achieved.

## 4.2 Continuous contrast variation in SAXS on PS-PMMA colloids

The measured scattering curves of the polystyrene particles are displayed in figure ???. In the region for  $q$  from  $0.03 \text{ nm}^{-1}$  to  $0.5 \text{ nm}^{-1}$  it is possible to observe the variation of the curve features corresponding to the particle form factor through the increase of the solvent electron density from  $333.7 \text{ nm}^{-3}$  at the top edge of the density gradient to  $360.3 \text{ nm}^{-3}$  at the maximum sucrose concentration. In this region, the experimental background is composed mainly by the contribution of the capillary scattering at the low  $q$ -region and the uniform scattering of the suspending medium. The experimental background scattering varies for different sucrose concentrations, but their variations are small and the background remains one order of magnitude below the sample scattering in the relevant Fourier region.

Upon increasing the solvent density, the position of the first minimum shifts from  $0.07 \text{ nm}^{-1}$  towards smaller  $q$ -values until it vanishes when the solvent electron density matches the average electron density of the measured particle. In the Fourier region of the scattering curves, several minima are observed which shift towards smaller  $q$ -values when increasing the solvent electron density. Upon subtracting the experimental background from the scattering curve, a decrease of the scattering intensity towards  $q = 0$  is observed only for the solvent electron density closest to the match point as depicted in figure ??. Therefore, background corrections can be neglected for systems with relatively high scattering power like in this study. For

low-scatterers, an accurate background correction by measuring the pure suspending medium at different sucrose concentrations might be required. The behaviour at low  $q$ -values will be further discussed in section ?? when evaluating the zero-angle intensity.

The presence of the clearly visible isoscattering point around  $q = 0.09 \text{ nm}^{-1}$  confirms the existence of an inner structure. This heterogeneous composition was previously reported for the same colloids by Minelli *et al.* (2014a), who observed methacrylic acid (MAA) and methylmethacrylate (MMA) at the particle surface, both monomer precursors of PMMA polymerization. A more detailed insight into the radial morphology is presented subsequently, using the theoretical framework already introduced.

## 4.3 Model dependent evaluation

### 4.3.1 Core-shell form factor fit

A core-shell model fit to the scattering curves is displayed in figure ?? for three representative contrasts. The model represents a radially symmetric particle, with a sharp interface between the outer shell and the inner core. This is a specific case of equation (2.2) with  $n = 2$

$$F_{CS}(q, R, R_{core}) = \Delta\eta f_{sph}(q, R) + \Delta\rho [f_{sph}(q, R) - f_{sph}(q, R_{core})], \quad (4.1)$$

where  $R$  and  $R_{core}$  are the outer shell and inner core radii respectively and the excess of electron density is  $\Delta\rho = \rho_{shell} - \rho_{core}$ . The simultaneous fitting of the form factor to the 40 measured scattering curves was performed by means of the method of least squares in the Fourier region Pedersen (1997). The calculated scattered intensity was modelled as the sum of the particle contributions and a two-component background  $I_{bg} = C_0 + C_4 q^{-\gamma}$ . The parameters  $\rho_{core}$ ,  $\rho_{shell}$ ,  $R$ ,  $R_{core}$  and  $\gamma$  were fitted simultaneously for all curves, whilst  $C_0$  and  $C_4$  were adjusted independently for each solvent density. A Gaussian size distribution was assumed. For the suspending medium electron density  $\rho_{solv}$  appearing in the contrast  $\Delta\eta$ , the value determined from the transmission measurement was used for each curve.

The obtained results are  $R = (49.7 \pm 2.8) \text{ nm}$ ,  $R_{core} = (44.2 \pm 0.9) \text{ nm}$ ,  $\rho_{core} = (339.7 \pm 0.1) \text{ nm}^{-3}$  and  $\rho_{shell} = (361.9 \pm 2.0) \text{ nm}^{-3}$ , which represent the radial structure of a dense, thin shell surrounding a lighter core, as seen in the inset of figure ?. The resulting average electron density of the particle is  $\rho_0 = (345.9 \pm 1.5) \text{ nm}^{-3}$  and the polydispersity degree,  $p_d = (22.8 \pm 6.0) \%$ . The best fitting background

corresponds to a value of  $\gamma = 4.3 \pm 0.5$ , close to the case  $\gamma = 4$  originating from large impurities or precipitates Pedersen (1994). The fit uncertainty was calculated with a confidence interval of one standard deviation.

It is noticeable that the calculated electron density of the core coincides exactly with the theoretical polystyrene electron density, although the electron density of the shell is remarkably lower than the theoretical value of  $383.4 \text{ nm}^{-3}$  for PMMA Ballauff (2001). This might arise from the lower density of the monomers used in the particle synthesis (MAA and MMA), which could have mixed with the styrene monomers resulting in a less dense material than PMMA. This model might present some differences with the real colloid system, as a diffusive interfacial layer can be expected between polymer phases in colloids Dingenouts *et al.* (1994b), especially for incompatible polymers such as PMMA and PS. On the other hand, the large quantity of scattering curves used for the fitting process and, accordingly, the decreased uncertainty suggests that the chosen sharp core-shell model has a great resemblance to the real particle.

## 4.4 Model-free approach to contrast variation data

### 4.4.1 Isoscattering point

#### Quantification: Relative standard deviation

The first isoscattering point is clearly visible in figure ?? . For a more quantitative evaluation, the relative standard deviation of the 40 measured curves at each  $q$  is calculated according to

$$\sigma_r(q) = \frac{1}{\bar{I}(q)} \sqrt{\frac{\sum_{i=1}^M (I_i(q) - \bar{I}(q))^2}{M - 1}}, \quad (4.2)$$

where  $\bar{I}(q)$  is the mean value of the intensity at  $q$  and  $M$  is the number of scattering curves. This value becomes minimal at an isoscattering point. In order to reduce the influence of outliers, a truncated mean value was utilized, disregarding the 10 % most dispersed data points. In figure ?? , the relative standard deviation is plotted as a function of the momentum transfer  $q$ , which shows several distinguishable minima corresponding to isoscattering points.

A precise determination of the isoscattering point positions is performed by fitting Lorentzian functions to the minima in the relative standard deviation plot, which allows the calculation of the model-free external radius of the particle by means of

Table 4.1: Experimentally determined position of the first five isoscattering points and the corresponding external particle radius  $R$ .

	$q^*$ (nm <sup>-1</sup> )	$R$ (nm)
$q_1^*$	0.0900	49.9
$q_2^*$	0.1516	51.0
$q_3^*$	0.2267	48.1
$q_4^*$	0.2822	49.9
$q_5^*$	0.3421	50.3

equation (2.6). The results are presented in table ???. The obtained particle radii vary in the range from 48.1 nm to 51.0 nm, although as predicted by Kawaguchi & Hamanaka (1992) for a polydisperse system, the isoscattering points get smeared out for larger  $q$ -values and the precision decreases, simultaneously with the increase of the solvent background at higher  $q$ -values. This can be directly observed in the quality of the experimental data, as the first two minima are clearly pronounced, while the subsequent minima appear smeared out. For instance, the isoscattering point  $q_5^*$  is already too weak for an accurate evaluation and the third minimum shows two remarkably close smaller minima which might affect the shape of the function. Therefore,  $q_1^*$  and  $q_2^*$  yield the most reliable values for evaluating the external radius of the particles, although all results are presented in table 4.1. The value derived from the isoscattering points  $R = 50.5$  nm differs by only 1.6 % from the radius calculated from the model fit in the previous section.

Due to the ambiguous definition of the isoscattering point diffuseness, a quantitative determination of the polydispersity of the suspended nanoparticles by means of the Lorentzian profile is rather challenging. Nevertheless, the narrow size distribution of the sample becomes clear by comparing the relative standard deviation values of the observed minima in figure ??? with a simulation using the structural parameters obtained in section §???. The value  $\sigma_r(q_1^*) = 0.11$  corresponds to a calculated ensemble polydispersity of 24 %. This value serves as an upper  $p_d$  limit due to the possible overestimation caused by the scattering contribution of the suspending medium.

#### 4.4.2 Guinier region

By analyzing the low  $q$ -region of the scattering curves, the so-called Guinier region, two important parameters can be obtained: the radius of gyration  $R_g$  and the intensity at zero angle  $I(0)$ . According to Feigin & Svergun (1987), the fit of equation

(2.8) to the Guinier region is mainly valid up to  $qR_g < 1.3$ . In this restricted  $q$ -range, too few data points are available for a reliable data analysis. Therefore, an extrapolation using the spherical form factor  $f_{sph}(q, R)$  over the range available before the first minimum has been employed instead to obtain  $R_g$  and  $I(0)$ .

As described in §??, the radius of gyration of a heterogeneous particle in a contrast variation experiment should behave according to equation (2.9). In figure ??, the experimental squared radius of gyration is displayed as a function of the suspending medium electron density. The best fit to the measured data with values  $\rho_0 = 343.7 \text{ nm}^{-3}$ ,  $\tilde{R}_{g,c} = 39.0 \text{ nm}$ ,  $\tilde{\alpha} = 4470 \text{ nm}^{-1}$  and  $\tilde{\beta} = 0 \text{ nm}^{-4}$  is shown by the solid line.

The positive value of  $\tilde{\alpha}$  validates the hypothesis that a more dense polymer (probably PMMA) surrounds a lighter core (PS) Stuhrmann (2008). The calculated average electron density of the particle  $\rho_0$  suggests a very thin layer of PMMA shell around the PS core, due to the proximity of its value to the polystyrene electron density ( $339.7 \text{ nm}^{-3}$ ). The value of  $\tilde{\beta} = 0$  proves a concentric model, where core and shell share the same centre. Using the same polydispersity value of 22.8 % obtained in the fitting process, the value for the particle shape radius of gyration results in  $R_{g,c} = 36.9 \text{ nm}$  and the external radius of the particle can be calculated assuming the particle as a spherical object ( $R_g^2 = \frac{3}{5}R^2$ ). This calculation gives  $R = 47.6 \text{ nm}$ , which is only 2.1 nm smaller than the calculated external radius  $R = 49.7 \text{ nm}$ , though it might be underestimated due to the choice of a possibly inflated polydispersity.

### Average electron density

Using the same set of 40 scattering curves, the behaviour of the zero-angle intensity under the contrast variation is also investigated by fitting equation (2.10) to the experimental  $I(0)$ , as depicted in figure ??. A minimum in the curve is observed at  $\rho_{solv} = 346.0 \text{ nm}^{-3}$ , which corresponds to the value of the average electron density of the particle. This value is in very good agreement with the result obtained by fitting the core-shell form factor. It is also noticeable that the minimum intensity is approximately 0, which means that the effective average density of the ensemble is equal to the average density of the particle Avdeev (2007a). This result further legitimates the previously made assumption that the ratio between the particle components' volumes is constant independent of the polydispersity and hence  $\tilde{\rho}_0 = \rho_0$ .

First point Comparison of accuracy

Extrapolatio Using just the Guinier region or extrapolating from first minimum



Table 4.2: Comparison of the different methods presented in this article to evaluate SAXS contrast variation data.

	$R$ (nm)	$\rho_0$ (nm $^{-3}$ )	$p_d$ (%)
Core-shell fitting	$49.7 \pm 2.8$	$345.9 \pm 1.5$	$22.8 \pm 6.0$
Isoscattering point*	50.5	-	$< 24$
Radius of gyration	47.6**	343.7	-
Zero-angle intensity	-	346.0	-

\*Mean value of  $q_1^*$  and  $q_2^*$

\*\*Using the polydispersity degree from the core-shell model fitting

## 4.5 Summary

Table 4.2 summarizes the results of all three presented methods. From the first two isoscattering points, values for the external radius and an upper bound to the polydispersity degree have been derived. Focusing on the Guinier region of the scattering curves, a value for the average electron density of the particles is found using the radius of gyration as well as the zero-angle intensity, the values of which differ by 2.3 nm $^{-3}$ . By fitting a core-shell model, an external radius of  $R = 49.7$  nm and an average electron density  $\rho_0 = 345.9$  nm $^{-3}$  have been obtained, which are in considerable agreement with the previous results, i.e., the values determined by the other methods are included in their confidence ranges, except for the  $\rho_0$  calculated with the radii of gyration.

From these results, it is evident that the radius of gyration interpretation produces the most deviant values. This might be founded in the complicated function fitted to the data and the reduced availability of  $q$ -range employed to obtain  $R_g$ .

The resulting polydispersity degree of the measured particles from the model fit is in agreement with the upper limit obtained with the radii of gyration. Nevertheless the polydispersity is the parameter determined with the largest uncertainty in the fitting process and therefore this result must be considered with care.

It can be concluded that the different approaches show consistent and complementary results about the size distribution of nanoparticles with radial inner structure, especially for the external radius of the particle and its average electron density. A precise value for the polydispersity degree could not be obtained as explained previously, although a credible upper limit to the polydispersity degree of 24 % could be given.

This article demonstrates that it is possible to perform continuous contrast variation for light nanoparticles by means of a density gradient and to collect a large

quantity of SAXS curves, which can be analyzed with complementary approaches to reveal a consistent insight into the size distribution and the inner structure of the suspended nanoparticles.

By means of a model-free analysis of the experimental data based on the isoscattering point theory, an average particle diameter of 101 nm was obtained. The analysis of the Guinier region of the scattering curves shows that the radial inner structure of the particles consists of a thin, more dense layer coating the polystyrene core. Complementing these results, a core-shell model fit showed that the core component of the particle had exactly the same electron density expected for polystyrene and the shell was composed of a compound with a density below that of PMMA. This core-shell structure was expected for chemical reasons due to the different hydrophobicity of PS compared to MMA and MAA.

Considering the similar electronic composition of these polymers and the average electron density of the particle  $\rho_0 = 346 \text{ nm}^{-3}$ , an average physical density of the particles of  $\rho = 1.07 \text{ g/cm}^3$  can be calculated. The precision in the determination of this density proves this technique as a useful tool and an alternative to other techniques like isopycnic centrifugation Vauthier *et al.* (1999); Arnold *et al.* (2006); Sun *et al.* (2009), widely used with biomacromolecules.

Nevertheless, future applications of this technique must consider the limited density range accessible with an aqueous sucrose solution, which restricts the applicability to light particles. More dense solutions prepared with heavy salts could be used as an alternative, but they might compromise the stability of the particles and lead to more complicated handling of the sample due to a decreased diffusion timescale. Other possible methods that vary the contrast of a single medium have already been proposed (e.g. ASAXS Stuhrmann (2007)), although a system fulfilling the requirements must be found and a large complementary dataset might be difficult to acquire.

## Chapter 5

# Simultaneous size and density determination of polymeric colloids

In the continuously growing world of nanotechnology, nanoparticles have a pre-eminent position, employed as pharmaceutical or cosmetic products(Guterres *et al.* 2007) and especially in the emerging field of nanomedicine. Indeed, nanoparticles open exciting new possibilities in this field as platforms for drug-delivery(Wang *et al.* 2012) or encapsulating imaging agents(Tao *et al.* 2011). Nowadays, polymeric colloids and biodegradable nanocarriers are finding many research and medical applications(Vicent & Duncan 2006) and are starting to undergo clinical trials(Patel *et al.* 2012; Beija *et al.* 2012; Cabral & Kataoka 2014).

The current advances in nanomaterial development for medical applications are focused towards tailoring polymeric nano-drug carriers with flexible surface functionalisation and controlled morphologies(Euliss *et al.* 2006; Yang *et al.* 2005). Size and shape, combined with the choice of polymer and the mechanical properties, are fundamental and defining aspects of the particle functions, e.g. their *in-vivo* biodistribution(Vittaz *et al.* 1996*b*; Mitragotri & Lahann 2009; Doshi & Mitragotri 2009) or their drug-delivery efficacy(Powers *et al.* 2006). Therefore, a full and consistent characterisation of all properties of nanoparticles is of crucial importance and must be carefully addressed.

The characterisation of polymeric nanoparticles remains a challenge due to their typically complicate internal structure(Beyer *et al.* 1990) and requires more than a single characterisation technique to detect these heterogenous compositions. For instance, electron microscopy is an effective tool for direct observation of the shape

and size distribution of nanoparticles, although it cannot conclusively elucidate their internal morphology.

The use of an ensemble-average and non-destructive technique such as small-angle X-ray scattering (SAXS) arises as an appropriate alternative (Leonard Jr *et al.* 1952; Motzkus 1959). SAXS can discern differences in the radial structure of polymeric colloids and offers advantages to other methods which require prior treatment of the sample and are not averaging (Silverstein *et al.* 1989*b*; Joensson *et al.* 1991*b*). Despite being a highly informative method for the accurate characterisation of polymeric particles, the difficulties in the interpretation of the scattering curves demands complementary experimental information (Mykhaylyk 2012).

The contrast variation method in SAXS varies systematically the electron density of the suspending medium by adding a suitable contrast agent, e.g. sucrose, in order to resolve the different contributions of the particle components to the scattering. By measuring SAXS patterns as a function of the adjusted contrast, a more detailed insight into the particle morphology can be obtained in comparison to single-contrast experiments (Bolze *et al.* 2004). For instance, the internal structure can be modelled in terms of the radial electron density (Dingenouts *et al.* 1994*a*, 1999*b*; Ballauff 2011*b*; Ballauff *et al.* 1996) and the individual contribution of each polymer can be distinguished (Beyer *et al.* 1990; Grunder *et al.* 1991, 1993; Ottewill *et al.* 1995; Bolze *et al.* 1997; Dingenouts *et al.* 1994*c*) as well as its density (Mykhaylyk *et al.* 2007).

The formation of a solvent density gradient within a capillary emerges as an intelligent strategy to measure SAXS patterns at a continuous range of contrasts and, as a result, collect in a relatively short timespan an extensive data set of complementary scattering curves (Garcia-Diez *et al.* 2015). The continuous contrast variation technique in SAXS is ideally suited for current synchrotron radiation sources, where high brilliance and collimation permit the measurement of the scattering curves within the diffusion time of the contrast agent.

This work demonstrates the simultaneous size and density determination using this technique with 3 polymeric particles of different sizes and polymeric species. By means of an aqueous sucrose density gradient, the measurements were achieved along a large range of suspending medium densities, from water density to that of poly(methyl methacrylate)’s, highlighting the relevance of the technique across a wide spectrum of polymers.

The article discusses the applicability of this method for the traceable size determination of these colloids, where a high-resolution size distribution of the particles is presented. Focusing on a low-density colloid, different evaluation approaches to SAXS contrast variation experiments are introduced and the advantages and drawbacks of a model-free formulation like the isoscattering point position are discussed,

together with the accuracy of the scattering shape factor.

In addition, a form factor model is fitted to the scattering curves to obtain decisive information about the internal morphology of the particle, which is not directly available by other techniques such as transmission scanning electron microscopy (TSEM), differential centrifugal sedimentation (DCS)(Fielding *et al.* 2012) or atomic force microscopy (AFM).

The ability of the continuous contrast variation technique to determine the density of polymeric colloids in suspension is also discussed. Normally, the density of the suspended particles can not be compared to the bulk density of the dry material. Such a complex question has been addressed by different methods, though with evident limitations. For example, the density of polymeric beads has been measured previously with field-flow fractionation (FFF) with high-accuracy but at the expense of *a priori* assumptions about the morphology of the particle(Giddings *et al.* 1981; Yang *et al.* 1983; Caldwell *et al.* 1986). Assuming the Stokes' diameter as the actual size of the colloid, recent advances in analytical ultracentrifugation allow the complementary characterisation of the size, density and molecular weight of gold nanoparticles(Carney *et al.* 2011).

The 3 polymeric colloids were also analysed by DCS and the results compared and discussed with those obtained by SAXS. DCS uses the sedimentation of particles through a density gradient to measure high resolution particle size distributions(Minelli *et al.* 2014*b*). Its accuracy typically depends on the knowledge of the density of the particles. When the size of the particle is known, DCS can alternatively be used to measure average particle's density.

Neumann(Neumann *et al.* 2013) used two sucrose gradients resulting in different viscosities and densities, where the altered settling velocity combined with linear regression analysis was used for the calculation of the size and density of silica nanoparticles and viruses. Bell(Bell *et al.* 2012) adopted a two gradient method based on the variation of the sucrose concentration to determine the density of the Stöber silica and the calibration standards used in DCS.

In this study, the size and density of low-density particles is independently determined by performing DCS measurements with two different discs using the sedimentation and flotation respectively of the particles through a density gradient and solving the relative Stokes' equations.

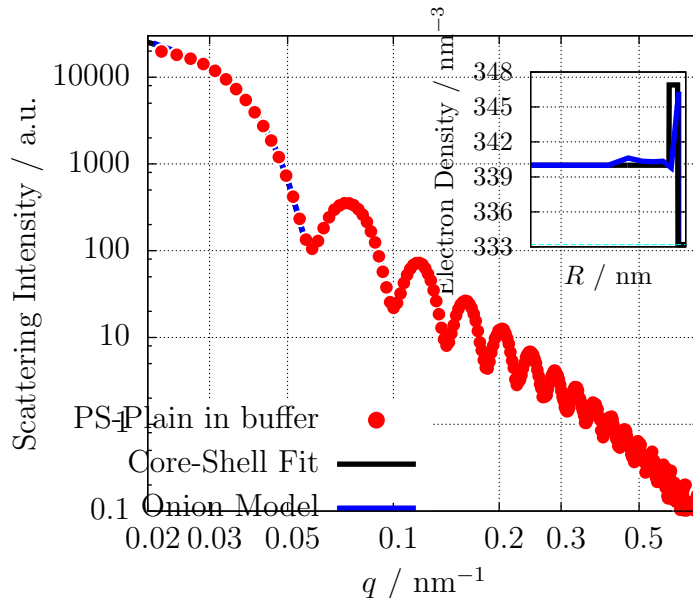


Figure 5.1: Scattering curve of PS-Plain in buffer. A core-shell and onion model fit to the experimental scattering curve are presented. In the inset, the electron density radial profile of these fits is shown, assuming the core is polystyrene with a density of  $339.7 \text{ nm}^{-3}$ .

## 5.1 Materials and methods

### 5.1.1 Particles and chemicals

Carboxylated polystyrene nanoparticles (PS-COOH) synthesized by multi-addition emulsion polymerization with a nominal size of 105 nm were purchased from Kisker Biotech (Steinfurt, Germany). Carboxylated poly(methyl methacrylate) colloids (PMMA-COOH) with a nominal diameter of 187 nm and plain polystyrene particles (PS-Plain) polymerized with  $< 1 \text{ wt}\%$  of a surface-active co-monomer with a nominal diameter of 147 nm were purchased from Microparticles (Berlin, Germany). Galden® PFPE SV90 was purchased from Solvay Plastics (Brussels, Belgium).

In figure 5.1, the SAXS curve of the PS-Plain particles in buffer at a single-contrast is shown. The large number of minima observed in the curve is remarkable and indicates the high monodispersity of the sample, which allows a traceable size determination of these colloids.

Upon trying different form factor fits detailed in the **Supplementary Information (SI)**, a simple core-shell structure with a sharp interface was found to be

Table 5.1: Parameters of the different DCS setups: composition of the sucrose gradients, average density of the gradients  $\rho_f$ , angular speed of the centrifuge  $\Omega$  and type of calibrant.

	Sucrose concentration (w/w)	$\rho_f$ (g cm <sup>-3</sup> )	$\Omega$ (rpm)	Calibrant
PS-COOH	from 2 % to 8 % in H <sub>2</sub> O	1.013	$2.0 \cdot 10^4$	A
PMMA-COOH	from 4 % to 12 % in H <sub>2</sub> O	1.025	$2.0 \cdot 10^4$	B
PS-Plain	from 2 % to 8 % in H <sub>2</sub> O	1.013	$2.4 \cdot 10^4$	B
PS-Plain*	from 4 % to 12 % in D <sub>2</sub> O	1.140	$2.4 \cdot 10^4$	C

\*Low density disc

the most suitable, suggesting a heterogeneous structure which is eluded by other characterization techniques, e.g. microscopy. The obtained particle diameter was  $(147.0 \pm 4.7)$  nm, where the fit uncertainty was calculated with a confidence level of one standard deviation ( $k = 1$ ) by examining the change in  $\chi^2$  when varying the diameter. The radial electron density profile of the core-shell fit is shown in the inset of figure 5.1, where a thin shell with high density surrounds a lighter core. This structure is likely due to the non-reacted monomers in the main matrix or the highly hydrophilic behaviour of the co-monomer, segregating polystyrene to the core.

### 5.1.2 Differential Centrifuge Sedimentation (DCS)

DCS measurements were performed with a CPS DC20000 instrument (CPS Instruments, Prairieville, LA, USA) upgraded to DC24000 for the PS-Plain measurements. The radial position of the detector was measured by injecting 100  $\mu$ L aliquots of water into the spinning disc initially empty until the accumulation of water produced a response in the detector. For the density gradient formation, the disc was filled with 14.4 mL of a sucrose (Amresco LLC, OH, USA) solution topped with 0.5 mL of dodecane to prevent evaporation. The detailed information of the gradients is summarised in table 5.1.

The measured turbidity at 405 nm was converted to the number of particles for each measured diameter by treating the particles as spherical Mie scatterers with no optical absorbance at the incident wavelength. Three different types of calibration particles were used: poly(vinyl chloride) colloids in water with density of 1.385 g cm<sup>-3</sup> and nominal size of  $(223 \pm 5)$  nm (calibrant A) and  $(239 \pm 5)$  nm (calibrant B) and polybutadiene colloids in 16 % sucrose mass fraction in heavy water with nominal size of  $(510 \pm 20)$  nm and density of 0.91 g cm<sup>-3</sup> (calibrant C). A standard disc configuration where the particles sediment through a lower density gradient

was used and additionally, a more recently developed set up which makes use of a disc where colloids float through a higher density gradient was also used for PS-Plain due to their low density (Fitzpatrick 1998). Measurements of PS-COOH and PMMA-COOH at 0.05 % w/v concentration were performed in triplicate. PS-Plain measurements were repeated seven times for each setup. Injection volumes were 100  $\mu\text{L}$ . Measurement uncertainties include both statistical and systematic uncertainty propagated from Stokes' equations.

### Stokes equations

The equation for the DCS is derived from the Stokes' law for a spherical particle of diameter  $D_p$  and density  $\rho_p$ :

$$D_p = \sqrt{\frac{18\eta \ln \frac{R}{R_0}}{(\rho_p - \rho_f) \omega^2 t_p}} \quad (5.1)$$

where  $t_p$  is the sedimentation time between radii  $R$  and  $R_0$  of the particle,  $\eta$  and  $\rho_f$  are the viscosity and the density of the fluid respectively and  $\omega$  is the disc angular frequency.

Before every sample measurement, a calibrant of known size  $D_c$  and density  $\rho_c$  is measured with the same set up for which the Stokes' law is also valid and can be expressed as:

$$D_p = D_c \sqrt{\frac{(\rho_c - \rho_f) t_c}{(\rho_p - \rho_f) t_p}} \quad (5.2)$$

### Combined analysis

When performing the measurements using two fluids, one with density  $\rho_L$  and one with higher density  $\rho_H$  (employed typically for lower density particles), one needs to solve the set of equations

$$D_p = D_{cH} \sqrt{\frac{(\rho_{cH} - \rho_H) t_{cH}}{(\rho_p - \rho_H) t_{pH}}} \quad (5.3)$$

$$D_p = D_{cL} \sqrt{\frac{(\rho_{cL} - \rho_L) t_{cL}}{(\rho_p - \rho_L) t_{pL}}} \quad (5.4)$$



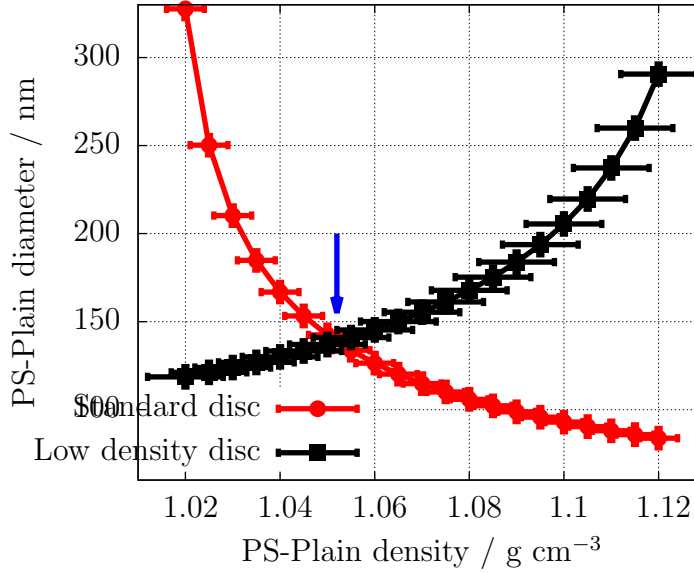


Figure 5.2: Dependence of the intensity-based modal Stokes' diameter on the particle density for PS-Plain particles analyzed in  $\text{H}_2\text{O}$ -sucrose (black) and  $\text{D}_2\text{O}$ -sucrose (red) gradients. The grey arrow indicates the crossing point of the data, which occurs for a diameter of  $(138.8 \pm 5.8)$  nm and a density of  $(1.052 \pm 0.010)$   $\text{g cm}^{-3}$

where  $cH$  and  $cL$  denote the calibrants used with high and low density fluids respectively and  $t_{pH}$  and  $t_{pL}$  are the sedimentation times of the particles measured in the high and low density fluids respectively.

The values of  $D_p$  and  $\rho_p$  which satisfy equations 5.3 and 5.4 can be found analytically or graphically. As depicted in figure 5.2, the two setups measure the same size and density of the colloid at the crossing point.

## 5.2 Technique validation for the determination of the particle size distribution

The morphology was further studied using the density gradient contrast variation technique described in §3 by varying the suspending medium electron density from 333.2 to 350.2  $\text{nm}^{-3}$ . By increasing the solvent contrast, the changes of the features in the scattering curves presented in figure 5.3a and the appearance of isoscattering points prove the multi-component composition of this colloid.

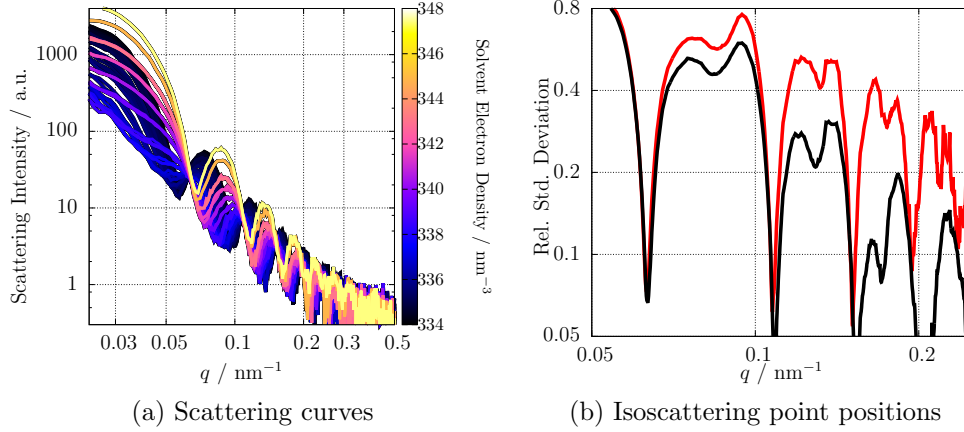


Figure 5.3: SAXS curves of PS-Plain obtained by density gradient contrast variation after solvent background subtraction. The inset shows the relative standard deviation calculated from all the scattering curves, where the minima correspond to the isoscattering points  $I_i$ .

From the 40 experimental scattering curves shown in figure 5.3a, a model-free size determination can be performed by locating the isoscattering points  $I_i$ , which are related to the radius  $R$  of the particle by  $\tan(q^*R) = q^*R$  (Kawaguchi *et al.* 1983).

The quantification of the isoscattering points positions  $q_i^*$  was performed by calculating the relative standard deviation of each  $q$  value across all the measured curves, as shown in the inset of figure 5.3a where the minima correspond to the fulfillment of the isoscattering condition.

The particle sizes obtained from the first 4 isoscattering points ( $I_1$  to  $I_4$ ) range between 142.4 and 144.4 nm, showing a good agreement for higher  $q$ -values as well. The precision of the isoscattering point positioning decreases for increasing  $q$  as demonstrated by Kawaguchi (Kawaguchi & Hamanaka 1992) and it is exemplified by the broadening of the minima for higher  $q$ .

The data can also be analyzed by using the so-called *shape factor* or *resonant term* which can be derived from the *basic functions* approach (Stuhrmann & Kirste 1965, 1967) described in the **SI**. The shape factor is defined by the scattering corresponding to the particle components impenetrable to the solvent, e.g. the external shape of the particle independently of its inner structure.

This approach is appropriate for many polymeric particles with a heterogenous morphology (Bolze *et al.* 2004), such as the PS-Plain colloid, because it enables the size distribution determination of the particles avoiding any *a priori* consideration about the particle composition.

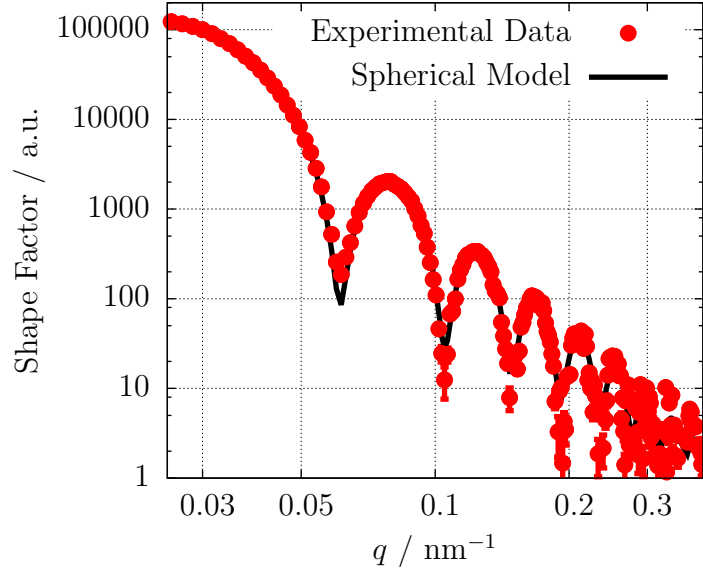


Figure 5.4: Experimental shape factor of the PS-Plain colloid calculated from 40 scattering curves and the spherical form factor fitted to the data.

The shape factor calculated from the measured scattering curves is depicted in figure 5.4 together with the spherical model fitted to the data, which employs a simple form factor that ignores the internal structure and a gaussian size distribution. From this fit, a mean particle size of  $(146.8 \pm 1.3)$  nm was determined. The fit uncertainty was determined as discussed before. By fitting an ellipsoid model to the shape factor, a sphericity of 98 % was obtained.

These results highlight the applicability of this technique for the characterization of the size and shape of polymeric colloids. Additionally, the associated uncertainty calculated with this approach is 3.5 times smaller than the one obtained with the single-contrast SAXS experiment.

### 5.2.1 Inter-laboratory comparison of the mean particle size

The improvement in the size accuracy with the shape factor approach is summarized in figure 5.5, where the size of the PS-Plain particles determined by different techniques in an inter-laboratory study is also presented (Nicolet *et al.* 2015, under review).

The figure compares the PS-Plain size measured by the ensemble techniques

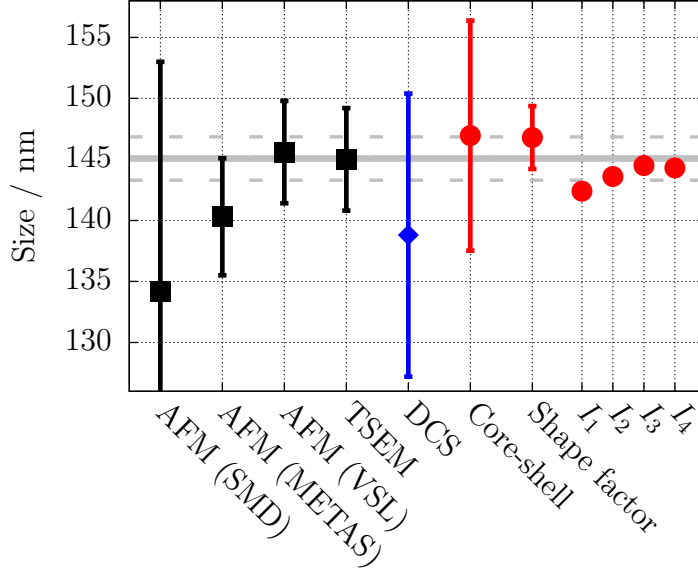


Figure 5.5: Comparison of the PS-Plain average size obtained with different techniques, where the errorbars correspond to the expanded uncertainty ( $k = 2$ ). The circles correspond to results obtained with SAXS and the diamonds to combined DCS measurements. The gray line defines the weighted mean of all the independent results.

SAXS and DCS and the imaging methods AFM and TSEM and presents the weighted mean value of all the independent results as a grey line, which corresponds to a diameter of 145.1 nm with an associated expanded uncertainty ( $k = 2$ ) of 1.8 nm. The SAXS results tend to larger values when modelling the scattering form factor, whilst the size obtained from the isoscattering points positions  $I_i$  present values slightly smaller than the calculated mean value. However, the maximum deviation from the weighted mean is less than 2 %.

The microscopy values are obtained from Belgian Service Métrologie-Metrologische Dienst (SMD), Swiss Federal Institute of Metrology (METAS) and Dutch Metrology Institute (VSL).

The DCS result is obtained by a combined analysis of two complementary centrifuge configurations as detailed in **SI**, where figure **S1** depicts the dependency of the measured particle size on the density values for the two setups. The two setups measure the same size and density at the crossing point of the data, which occurs for a diameter of  $(138.8 \pm 5.8)$  nm and a density of  $(1.052 \pm 0.010)$  g cm<sup>-3</sup>. The measured size fits within its uncertainty in the confidence interval of one standard deviation of the inter-laboratory comparison.

All the techniques are in very good agreement, even considering that they are based on different physical principles. The improvement in accuracy for the size determination with SAXS by using the shape factor approach is further sustained by this comparison.

This improvement was confirmed by employing the same approach with the PS-COOH colloids. The size obtained from the core-shell model fit is  $(99.4 \pm 5.6)$  nm (Garcia-Diez *et al.* 2015), while the value obtained from the shape factor calculation is  $(101.4 \pm 2.4)$  nm. Again, the uncertainty associated to the size decreases by  $\sim 60$  %, whilst it is still in accordance with the size obtained with the isoscattering points positions of 101.0 nm with a standard deviation of 1.1 nm.

Due to the low polydispersity of the PMMA-COOH particles and their homogeneous composition, a spherical form factor fit to the single-contrast scattering curve provides already a very accurate size  $(186.5 \pm 2.3)$  nm. In this case, contrast variation experiments in SAXS show no advantages.

It has been demonstrated that the possibility to determine the particle size distribution by the scattering shape factor is a clear improvement to single-contrast SAXS techniques reducing relevantly the uncertainty, although an accurate determination of the contrast and a relatively high number of scattering curves are required.

Nevertheless, another contrast variation evaluation approach such as the isoscattering points presents as well certain assets which can not be ignored. For instance, the independence of  $q^*$  from the sample contrast facilitates its easy application. On the other hand, the diffuseness of the isoscattering point position due to the polydispersity and ellipticity of the sample (Kawaguchi & Hamanaka 1992) arises as an indisputable drawback.

## 5.2.2 Colloidal size distribution

An important attribute of polymeric colloids is their polydispersity, as the suitability for specific applications depends on their spread in size. For example, colloids are known to induce different inflammatory responses depending on their size (Kusaka *et al.* 2014). The polydispersity degree  $p_d$  is calculated as the full width at half-maximum of the number-weighted particle size distribution divided by its average value.

The SAXS results determine a  $p_d$  for the PS-Plain colloids of 6.1 %, which is an indicator of a very monodisperse distribution, as also suggested by the regular minima observed in figure 5.1. Particle polydispersities measured by DCS are also low as observed in figure 5.6, ranging from 7.8 % measured with the standard set up, to 11.3 % measured with the low density disc setup. The standard setup appears therefore

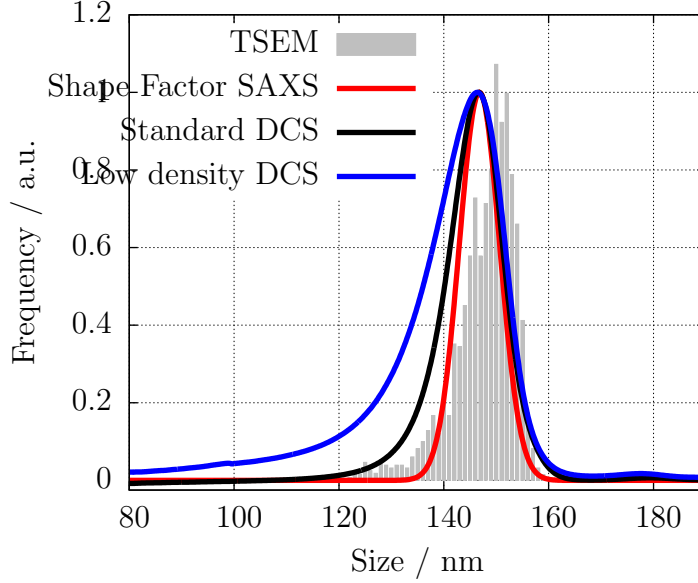


Figure 5.6: Number-weighted size distribution of PS-Plain particles measured by DCS, TSEM [49] and SAXS with the scattering shape factor approach.

to achieve a higher resolution size distribution. The size distribution measured by TSEM with a  $p_d$  of 8.3 % shows good agreement with the ensemble techniques.

The measurements obtained by AFM provide polydispersity degrees larger than 10 % (Nicolet *et al.* 2015, under review) and, therefore, slightly broader size distributions than those calculated by SAXS, TSEM and standard DCS. This can be in part attributed to the low statistics that typically affect imaging methods, along with artefacts associated with the posterior analysis.

For instance, in the TSEM images (Nicolet *et al.* 2015, under review), smaller and larger populations with different contrasts have been observed which could affect the evaluation of the density measured by ensemble techniques in §5.4, as the particle average density might vary. Indeed, when a bimodal distribution is used to analyze the SAXS shape factor of PS-Plain particles, a second size population is found at 101 nm in agreement with TSEM, while the main mode maintains a  $p_d$  of ca. 5 %.

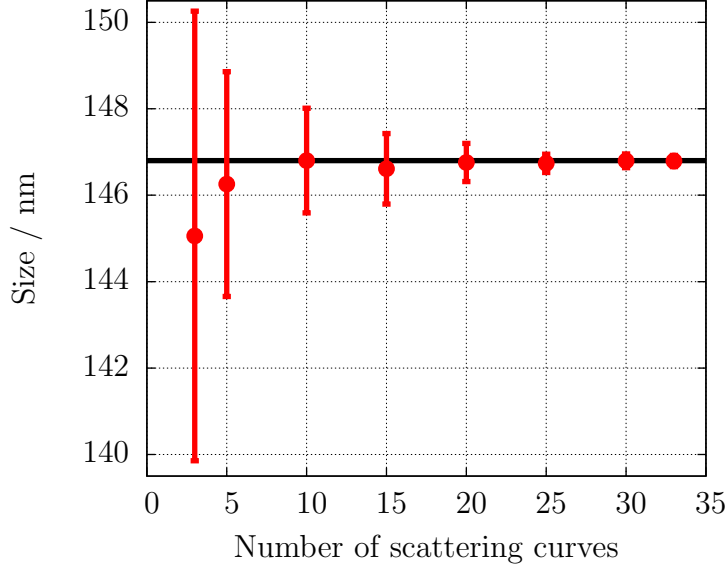


Figure 5.7: Size of PS-Plain as a function of the number of scattering curves used in the shape factor calculation.

## 5.3 Considerations about contrast variation data evaluation

### 5.3.1 Shape factor formalism

The shape factor obtained by density gradient contrast variation has been demonstrated as a powerful technique which can provide precise information about the size distribution and shape of the colloid by fitting a simple form factor.

However, an accurate determination of the suspending medium density for each scattering curve is required, due to the increased uncertainties Lefebvre *et al.* (2000) that can arise from the resolution of the system of linear equations described in the **Supplementary Information**.

Besides, a minimum of 3 scattering curves measured at different contrasts is necessary to obtain the resonant term, although an increasing number improves the determination of the size distribution. This issue has been addressed with the data measured by the density gradient contrast variation of the PS-F colloid.

From the 40 experimental curves, only a limited number  $N$  was randomly selected to compute the shape factor, while this process was repeated 100 times. The mean

	Raw data		Corrected data		Deviation
	$q^*$ (nm <sup>-1</sup> )	Size (nm)	$q^*$ (nm <sup>-1</sup> )	Size (nm)	%
$q_1^*$	0.0633	142.0	0.0631	142.4	0.3
$q_2^*$	0.1088	142.0	0.1076	143.6	1.1
$q_3^*$	0.1537	141.9	0.1510	144.4	1.7
$q_4^*$	0.206	136.6	0.195	144.3	5.3

Table 5.2: Isoscattering points position and the corresponding particle size for the scattering curves before and after background correction

size obtained from this data set and its standard deviation are plotted in figure 5.7 as a function of  $N$ .

The effect of increasing the number of measured contrasts evidences that the result tends asymptotically to the value of 146.8 nm discussed in §5.2 and the standard deviation of the 100 iterations decreases for large  $N$ , e.g. the associated uncertainty. This outcome emphasizes further the advantages of the continuous contrast variation technique due to the large number of scattering curves at different contrasts which can be easily measured.

Simulation depending on number of curves

## Advantages and disadvantages

### 5.3.2 Isoscattering point approach

It is noticeable in figure 5.3b that the subtraction of the suspending medium scattering plays an important role in the  $q^*$  values as the curve shifts to smaller  $q$ -values when subtracting the previously calibrated solvent scattering, as summarized in table 5.2.

The isoscattering point  $q^*$ , where all the scattering curves have the same intensity independently of the contrast, was first formulated by (Kawaguchi *et al.* 1983). It relates in a simple way the position of  $q^*$  with the size of the particle inaccessible to the suspending medium and, thus, a good method to determine the diameter of the colloid.

The theory defines  $q^*$  as a morphological parameter independent of the suspending medium density, which is a enormous practical advantage as it can be located without the proper calibration of the contrast. In cases where the composition of the buffer is unknown or the density of the solvent cannot be properly calibrated, the isoscattering point position might still be quantifiable by calculating the relative standard deviation of all the measured scattering curves. In order to obtain reliable



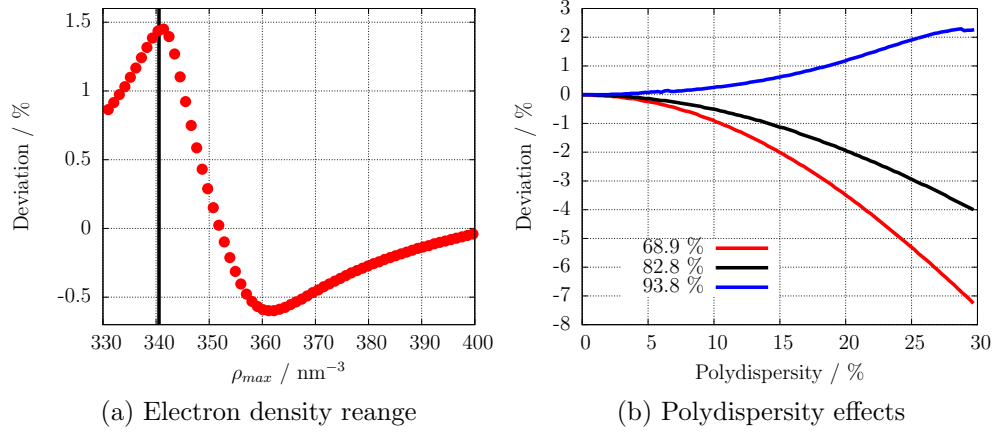


Figure 5.8: Deviation of the size calculated from  $q_1^*$  from the nominal value depending on the contrast range ( $330 \text{ nm}^{-3}$ ,  $\rho_{max}$ ) or the polydispersity of the core-shell colloid.

results, a proper subtraction of the solvent scattering must be performed as discussed in §5.2.

Nevertheless, it has been demonstrated that the polydispersity of the latex and its ellipticity influence the position and diffuseness of  $q^*$ , principally at high  $q$ -values [REF](#). This can disturb the size determination for lattices with broad size distributions and limit the applicability of this technique.

Besides, this work demonstrates that the  $q^*$  value determined with the previously described method depends on the range of solvent densities used in the contrast variation experiment. This conflicts partly with the initial intuition that this technique is independent of the experimental practice, although it can be avoided by selecting the range skillfully.

For this purpose, it was simulated the result of a contrast variation experiment with 10 different solvent densities for a colloid with the morphology and size distribution obtained with the onion model and presented in figure 5.1. Using a lower bound to the contrast range of  $\rho_{min} = 330 \text{ nm}^{-3}$  and increasing systematically the upper limit, it is shown in figure 5.8a that the calculated result deviates from the nominal value until 1.5%. This could be one explanation behind the size differences observed in figure 5.5 between the SAXS results.

In this example, the largest deviations occur when the contrast range excludes the average density of the latex, e.g. match point (depicted as a grey line), or when  $\rho_{max}$  is close to this matching density. These observations are applicable to other contrast variation experiments, advising to include the match point in the contrast

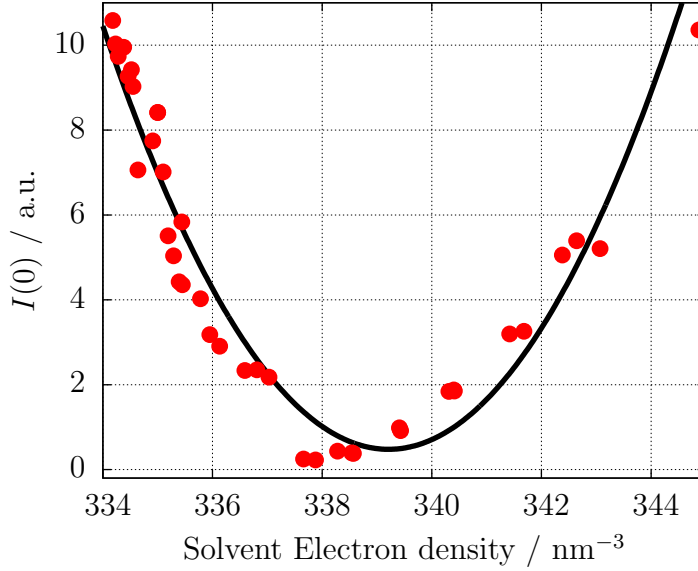


Figure 5.9: Intensity at zero-angle of PS-Plain particles as a function of the solvent electron density measured with continuous contrast variation in SAXS. The minimum defines the average electron density of the particle.

range and extending it along all the possible components' densities of the colloid.

#### Advantages and disadvantages

### 5.4 Determination of the particle physical density

In contrast variation SAXS, the solvent electron density which matches the average electron density of the particle ( $\rho_0$ ) corresponds to a minimum in the intensity of the scattering curve. In order to quantify the particle density, the scattering intensity of PS-Plain at zero angle  $I(0)$  is examined along the contrast range of the experiment as shown in figure 5.9. The value of  $I(0)$  was determined by extrapolation to  $q \rightarrow 0$  using a spherical form factor function fitted to the available range before the first minimum.

This parameter behaves parabolically around the average electron density of the particle like  $I(0) \propto (\rho_0 - \rho_{solv})^2$  (Avdeev 2007b). From the position of the minimum,  $\rho_0$  can thus be solved. The parabolic fit to the data is plotted as a black line in figure 5.9 and results in  $\rho_0 = (339.2 \pm 1.0) \text{ nm}^{-3}$ , which is consistent with the tabulated

value of dry bulk polystyrene  $339.7 \text{ nm}^{-3}$  (Dingenouts *et al.* 1999b).

The electron density is directly proportional to the physical density. Nevertheless, an assumption about the polymer (or monomer) components and their atomic structure is necessary for the calculation. Therefore, a typical value of  $Z/A = 0.54$  was adopted, where  $Z$  and  $A$  are the average atomic number and mass of the polymer respectively.

### 5.4.1 Validation through comparison with DCS

In figure 5.10, the value of  $(1.043 \pm 0.003) \text{ g cm}^{-3}$  obtained with the  $I(0)$  approach from the continuous contrast variation experiment is compared to the average density of the PS-Plain colloid measured with different DCS configurations. For single disc setups, the size value used for the density calculation was 147 nm, as measured by SAXS, while combining the information from the two setups allowed the measurement of the density independently of the particle diameter, as explained in §5.2.

The results agree with each other within their stated measurement uncertainties, although DCS measurements exhibit slightly higher densities than SAXS. Typical causes of systematic errors in DCS are the inaccuracy of the size and density of the calibration standard and the thermal variation in the centrifuge gradient during the measurements, which affect its viscosity and density (Kamiti *et al.* 2012). A temperature variation within the gradient of about 7 degree C before and after measurements was detected and a period of 30 min was considered appropriate to reach reliable thermal equilibrium. In the low density disc configuration, the accuracy of the average density of the D<sub>2</sub>O sucrose gradient becomes an important source of uncertainty.

### Uncertainties

In SAXS, the uncertainty is associated to the vertical size of the focused X-ray beam as in (Garcia-Diez *et al.* 2015). Furthermore, the result can be affected by the polymeric composition of the colloid, and therefore, the assumption of  $Z/A$ .

### 5.4.2 Use for homogenous polymeric colloids

The applicability of the continuous contrast variation techniques is further discussed by comparing with DCS for higher-density polymeric colloids, as summarized in figure 5.10. The density of the PS-COOH particles derived from the  $I(0)$  approach is in excellent agreement with that measured by DCS using a standard configuration

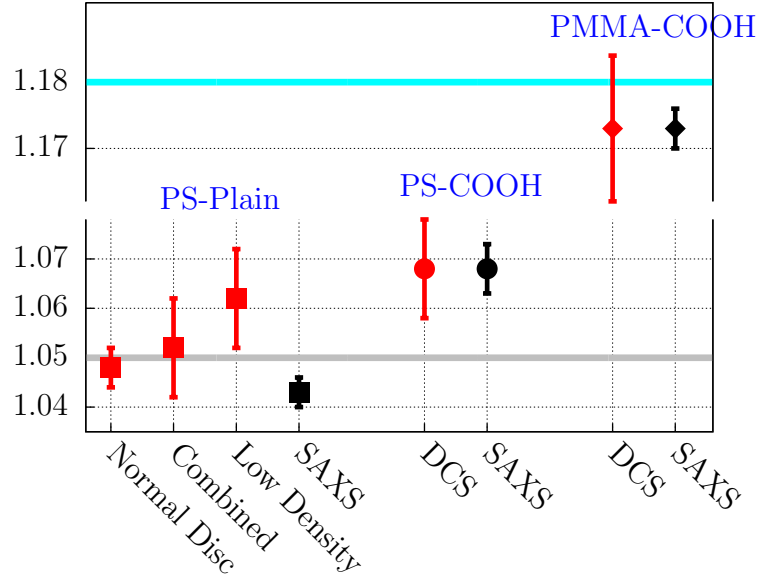


Figure 5.10: Comparison between the physical densities of 3 polymeric colloids measured with SAXS using the  $I(0)$  approach and DCS: PS-Plain (squares), PS-COOH (circles) and PMMA-COOH (diamonds). The nominal densities of polystyrene ( $1.05 \text{ g cm}^{-3}$ ) and PMMA ( $1.18 \text{ g cm}^{-3}$ ) are also shown in the plot as horizontal lines [22].

and assuming a particle diameter of 99.4 nm, which was obtained by SAXS. These core-shell particles, more dense than polystyrene (Garcia-Diez *et al.* 2015), illustrate the tendency during the emulsion polymerization to segregate polar and nonpolar components (Dingenouts *et al.* 1994c).

Similarly, the density of the PMMA-COOH colloids was measured using the standard DCS setup and assuming a diameter of 186.5 nm, as measured by SAXS. This value is compared to the density obtained by computing the intensity at zero-angle of a continuous contrast variation experiment. Again, both techniques are in excellent agreement and reveal a physical density slightly lower than the expected PMMA density of  $1.18 \text{ g cm}^{-3}$  (Dingenouts *et al.* 1999b).

This result highlights the fact that the density of polymeric colloids in suspension may vary from that of bulk materials, for example dry particles. For instance, a volume variation can be expected when going from the MMA monomer to the polymer PMMA (Nichols & Flowers 1950) which might reduce the colloid density.

## 5.5 Summary

This work demonstrates how continuous contrast variation in SAXS emerges as a powerful characterisation technique for polymeric colloids, which can determine their size and density in a traceable way. For instance, the accuracy in the density information achieved with the density gradient technique is remarkable and extends along a rather large density range of polymers.

Since contrast variation in SAXS is very sensitive to small electron density differences in the colloid morphology, the applicability of this method to investigate the inner structure of 3 different particles has been discussed. This is of paramount importance in polymeric particle characterisation because the direct observation by imaging techniques is inadequate for this purpose.

The detection of core-shell structures in polymeric colloids appears as essential for understanding the possible processes occurring during the particle formation, e.g. the consequences of emulsion polymerization synthesis.

These results were compared successfully with other techniques. In particular, SAXS measurements of the density of these colloids are in excellent agreement with those performed by DCS. The use of a novel DCS setup is also shown, which makes use of a centrifuge disc where the colloids float through a gradient of higher density, in contrast to a standard setup where the particles typically sediment. The use of the two complementary DCS configurations allowed the simultaneous determination of both the size and density of polymeric colloids consistently with the SAXS results.

Furthermore, different evaluation approaches to contrast variation SAXS data are examined in detail. The isoscattering point framework is found to be of easy utilization and very appropriate for spherical and quite monodisperse colloids. On the other hand, the calculation of the scattering shape factor arises as a precise sizing technique which can additionally provide an insight into the particle shape, although a high number of measurements with different contrasts and an accurate calibration of the system are required.

With the continuous contrast variation technique in SAXS, a more precise characterisation of the morphology of polymeric particles is achieved which opens new opportunities to investigate complex polymeric colloids. Besides, both ensemble techniques presented in this paper arise as powerful methods which can describe simultaneously the density and size distribution of polymeric colloids at the nanoscale.

# Chapter 6

## Continuous contrast variation applied to relevant bio-materials

### 6.1 Materials and methods

#### 6.1.1 Caelyx: PEGylated liposomal doxorubicin

#### 6.1.2 Iso-osmolar contrast agent: Iodixinol

#### 6.1.3 Sterically Stabilized Liposomes (SSLs) of different sizes

#### 6.1.4 Lipoproteins

HDL and LDL

## **6.2 Traceable size determination of a liposomal drug**

### **6.2.1 Isoscattering point approach**

### **6.2.2 Shape factor calculation**

## **6.3 Osmotic effects in liposomes**

### **6.3.1 Application to drug-stabilized liposomes**

### **6.3.2 Size dependency of the osmotic activity**

## **6.4 Application to blood plasma components**

### **6.4.1 HDL**

### **6.4.2 LDL**

### **6.4.3 Literature comparison**

## **6.5 Protein-coated low-density nanoparticles**

### **6.5.1 Single-contrast SAXS**

Caterina Minelli Paper ECASIA

### **6.5.2 Contrast variation**

Isopoint subtraction, as in BioSurf

## **6.6 Summary**

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