

# CRYSTALS workshops

## “twins”

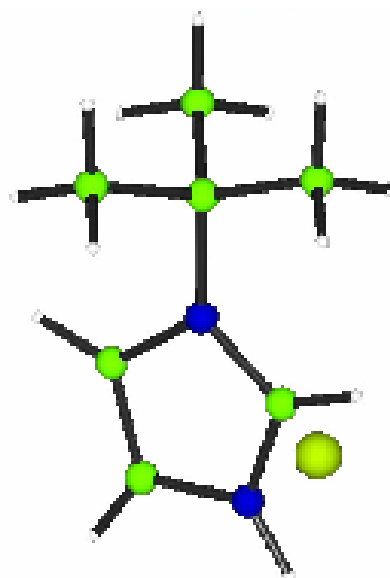
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### Keen ( $C_7H_{13}ClN_2$ ): Pseudo-merohedral twin

#### Summary

This data for this material were collected by on a Nonius KappaCCD diffractometer with Mo  $K\alpha$  radiation.

Expected Formula  $C_7 H_{13} Cl N_2$ . Temperature 150K  
Monoclinic b unique



- The structure was solved with 2 molecules in the asymmetric unit in  $P2_1$  but failed to refine to an R-factor below 30%.
- Using the systematic *weaknesses* as a guide, it was postulated that the sample was a poor quality in  $P 2_1/c$ . The structure again solved in this space group, but failed to refine below 20%.
- Finally, it was recognised that the material was twinned. The R factor fell to below 3%, with all the hydrogen atoms showing in a difference map.

#### First Attempt (for your information only):

Importing and solving using the space group  $P 2_1$  gives a reasonable structure with two independent molecules, but it fails to refine.

SIR92 leaves a diagnostic file in the working folder. CAMERON can also be used to get a global view of the structure.

#### Tools at your disposal:

You could open *SIR92.lis*. Type: **\$notepad sir92.lis**

The volume per atom looks OK, there is no translational pseudo-symmetry, but all the statistics point strongly to acentric. This is unusual (but not impossible) for a synthetic material.

You could use **CAMERON** to look along the *b* axis at a ‘complete’ packing diagram. The two independent molecules might be related by a *c* glide. To do this you would:

Click **Cameron** on the toolbar. Change selection from **Unpack** to **Complete**. Click the button **Axis b**. Close Cameron window. **Do not** apply changes made in Cameron.

#### Second Attempt (for you to try):

If you have tried the **First Attempt**, navigate to the **Keen** folder and delete the **crfilev2.dsc** file. Import the data (*keen.ins*, *keen.hkl*), trying space group **P 1 2<sub>1</sub>/c 1**. Follow the same steps as for the other structures, but take note of the statistics for the systematic absences, and for the merging R factor.

The structure should solve in Sir 92. **Delete the spurious C peaks, but be careful not to delete the Cl anion!**

You will find that it still fails to refine to a decent R-factor.

#### Diagnosis:

Scroll backup up the text page until you find the reflection input section.

- The mean value for the systematic absences is unusually high (it is almost always greater than zero).
- $R_{int}$  is good, especially for the medium and strong reflections.

Centro-symmetric space-group.  
 \* Friedel Pairs will be merged.  
 \* Removing systematically absent reflections

119 absences, mean Fo = 23.090 rms Fo/sigma(Fo) = 24.535

Fo range	-ve	1.000	2.000	4.000	8.000	16.000	Remainder
Mean Fo	-3.675	0.000	1.620	3.166	5.384	12.344	59.662
Number	29	0	6	17	17	6	44

Fo/sigma range	-ve	1.000	2.000	4.000	8.000	16.000	Remainder
rms (Fo/sigma)	1.991	0.581	1.370	2.828	5.875	12.295	44.714
Number	29	14	14	9	11	7	35

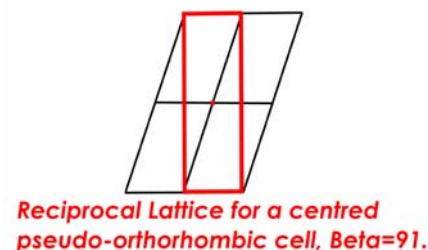
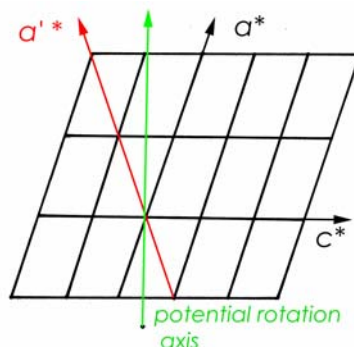
3794 reflections accepted 119 reflections rejected

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 2016 merged reflections output  
 1857 Reflections greater than 3 sigma(i), 92.113 percent of data  
 Rint = [Sum(|Fsq-<Fsq|)/sum(Fsq)] = 0.018  
 Rmerge = SQRT[sum[w(Fsq-<Fsq)\*\*2] / sum[w<Fsq\*\*2]] = 0.013  
 Rint for I>10sigma, 10sigma>I>2sigma, I<2sigma  
 0.015 0.074 0.858  
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- Look at the systematic absences plot

Start worrying about the crystal being twinned. A scale drawing of the reciprocal lattice shows the possibility of a pseudo-orthorhombic super-cell (in red).



The pseudo-orthorhombic super lattice means that twinning is possible by rotation about either  $c^*$  or  $a$ .

Rotation of the whole lattice about either axis does not result in exact overlap of reciprocal lattice points - this type of twinning is termed pseudo-merohedral (TLQS) twinning. However, the almost orthorhombic  $\beta$  angle of 91.13 degrees means that in practice both lattices overlap, so that every observation contains contributions from both twin domains.

This type of twinning is easily treated in CRYSTALS – once the twin law is identified. From the diagram we propose a rotation of 180 degrees about  $c^*$ , which transforms the indices as:

$$h' = \begin{bmatrix} -1 & 0 & 0 \\ 0 & -1 & 0 \\ 1 & 0 & 1 \end{bmatrix} . h$$

ROTAX will do this for you.

### Twinned Refinement: Easy

Guided treatment of two-component merohedral and pseudo-merohedral twins is implemented in the CRYSTALS GUI: Choose

**GUIDE > Analyse > Rotax Analysis > ROTAX**

Select a low "fom" twin law and apply it.

Use the GUIDE to set up a refinement. Note that the twin elements are now automatically included.

The R-factor should drop very quickly, and at the end of the guided refinement should end up below 3%

