

Q.1. The plastic behaviour of a metal can be expressed as : $\sigma = 500 \epsilon^{0.5}$ MPa.

Estimate the strength of a bar of this material if it is uniformly cold-worked to a reduction of $r = 0.3$. (reductn in area of c/s)

$$\rightarrow r = \frac{A_0 - A}{A_0} = 0.3 \quad , \quad \epsilon = \ln\left(\frac{l}{l_0}\right)$$

$$A_0 - A = 0.3 A_0 \quad = \ln\left(\frac{1}{1-r}\right)$$

$$A = 0.7 A_0 \quad = \ln\left(\frac{A_0}{A}\right)$$

$$\frac{A_0}{A} = \frac{10}{7}$$

$$\therefore \epsilon = \ln\left(\frac{10}{7}\right)$$

Q.2. (a) Strain hardening behaviour of a low carbon steel is given by : $\sigma = 700 \epsilon^{0.2}$ MPa
Estimate strength of bar if it is cold-worked to 50% reduction.

(b) Suppose another bar of the same steel was cold-worked ~~an~~ unknown amount and then further cold-worked 15% more. and found to have Y.S. of 525 MPa.
Unknown amt. of cold work = ?

$$\rightarrow (a) \frac{A_0 - A}{A_0} = 0.5$$

$$0.5 A_0 = A$$

$$\therefore \frac{A_0}{A} = 2 \quad \Rightarrow \quad \epsilon = \ln\left(\frac{A_0}{A}\right) = \ln 2 \rightarrow \sigma = 650 \text{ M}$$

$$(b) 525 = \epsilon_1 + \sigma \ln\left(\frac{1}{1-0.15}\right)$$

$$= \epsilon_1 + 700 \left(\ln \frac{100}{85} \right)^{0.2} \times$$

$$\epsilon_{\text{tot}} = \left(\frac{\sigma}{k} \right)^{1/n} = 0.237$$

$$\epsilon_{\text{known}} = 0.1625$$

Q.3. A cylindrical test specimen of dia = 10mm and gauge length = 50mm is extended to 65mm. Determine true strain. If ultimate tensile strain occurs @ F = 25000N and extension of 70mm. Determine strain hardening exponent 'n' and UTS of material.

$$\rightarrow \text{(A) True strain } = \epsilon = \ln \left(\frac{l_0}{l} \right)^t = \ln \left(\frac{50}{65} \right)^t = 0.262$$

$$\text{(B) } \Delta l = 70\text{mm}$$

$$\epsilon = \ln \frac{l}{l_0} = \ln \frac{70}{50} = 0.336 = n$$

$$\sigma_{\text{true}} = \frac{F}{A} = \frac{25000}{\pi r^2} = 318.47 \text{ MPa}$$

$$\sigma_{\text{true}} = \frac{25000}{\pi r^2} \left(1 + \frac{20/70}{2364} \right) = 445.2 \text{ MPa}$$

Q.4. The aluminium test specimen 100mm long, 20mm wide, 2mm thick is elongated to 130mm. If anisotropy ration 'r' value is 2. Determine $\epsilon_x, \epsilon_w, \epsilon_t$

$$\rightarrow \epsilon_x = \ln \frac{l}{l_0} = \ln \frac{130}{100} = 0.26$$

$$\frac{\epsilon_w}{\epsilon_t} = 2 \quad \epsilon_x + \epsilon_w + \epsilon_t = 0$$

$$\epsilon_x = -3\epsilon_t$$

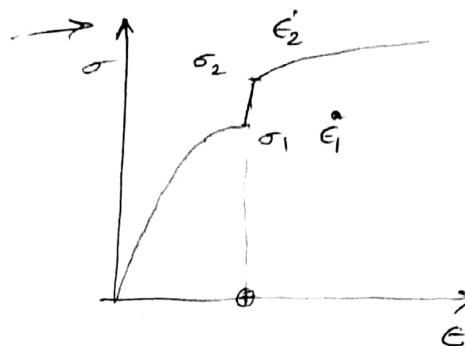
$$\epsilon_t = \frac{-1}{3}\epsilon_x = -0.0874$$

$$\epsilon_w = \frac{-2}{3}\epsilon_x = 0.195$$

Q.5 K, n, m value of stainless steel material
 $\downarrow \quad \downarrow \quad \downarrow$
 $1140 \text{ MPa} \quad 0.35 \quad 0.01$

A test piece has initial width ~~and~~ thickness and gauge length are $12.5, 0.45, 50 \text{ mm}$ resp.

Determine increasing load when extension is 10% and extension rate of gauge length is $0.5 - 50 \text{ mm/min}$.
~~is increased from~~



$$\sigma = K \cdot \epsilon^n \cdot \dot{\epsilon}^m$$

$$\dot{\epsilon}_1 = \frac{U_1}{L}$$

$$\frac{L - 50}{50} = \frac{10.5}{100 \times 2} \rightarrow L = 55 \text{ mm}$$

$$\therefore \epsilon = \ln \frac{l}{l_0} = \ln \left(\frac{55}{50} \right) = 0.0953$$

$$\dot{\epsilon}_1 = \frac{0.5}{50 \times 60} = 1.515 \times 10^{-4}$$

$$\dot{\epsilon}_2 = \frac{50}{50 \times 60} = 0.01515$$

$$\sigma_1 =$$

$$\sigma_2 =$$

$$\Delta \sigma = \sigma_2 - \sigma_1$$

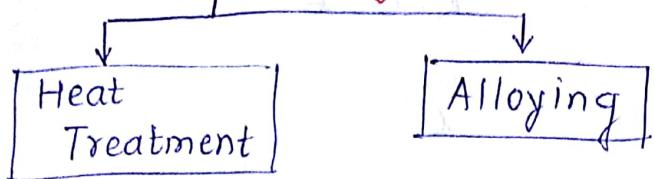
$$\Delta P = \Delta \sigma \times \frac{A_0 l_0}{l}$$

FORMING ...

01/08/2018

Heat Treatment Process :—

- ① Annealing : Furnace cooling
- ② Normalising : Air cooling
- ③ Hardening : Water cooling / oil
- ④ Case-Hardening : Only surface gets hardened.



- * Diffusion-based : wt % of phase = $f(\text{Temp}, \text{time})$
- * Non-diffusion : Austenite \rightarrow Martensite
[Lower the temp., higher the prod. conc.]

- ① Flame-hardening
- ② Induction-hardening (Placed in mag. field)
- ③ Laser hardening

- ⑤ Tempering : [Hardness / Strength / Ductility / Impact strength]

DEFORMATION :—

Temporary

$$\epsilon = f(\sigma) \quad [\text{purely elastic}]$$

(Temporary) \hookrightarrow Hooke's Law

$$\epsilon = f(\sigma, t)$$

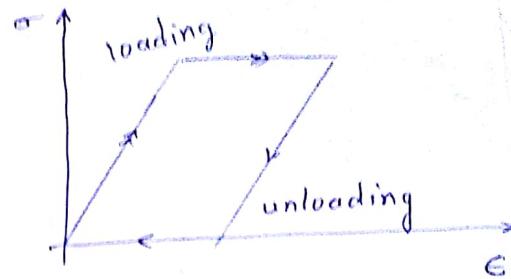
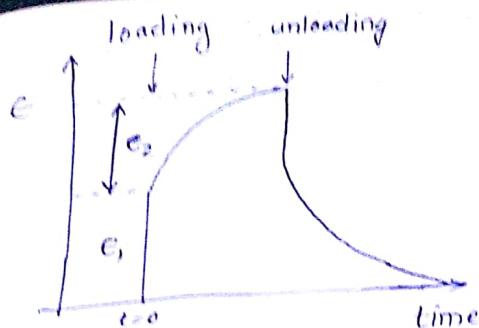
[comes back to original after some time]

An-elastic deformation...

Permanent

$$\epsilon = f(\sigma, T, \dot{\epsilon})$$

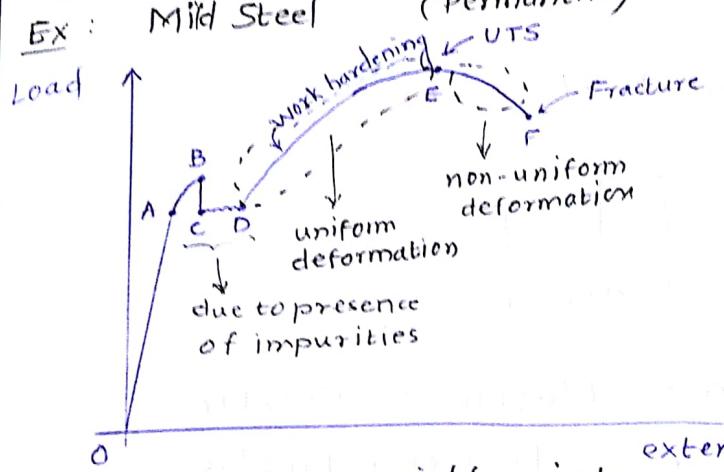
07/08/2018



$$\epsilon_{\max} = \epsilon_1 + \epsilon_2$$

TEMPORARY

Ex : Mild Steel (Permanent)



* Engg-stress-strain diagram ...

$$e = \frac{L - L_0}{L_0} = \frac{\Delta L}{L_0}$$

$$s = \frac{P}{A_0}$$

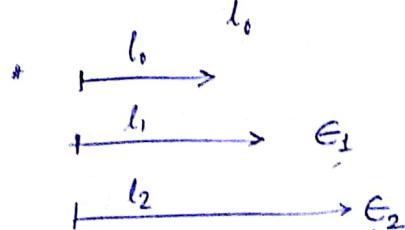
B - Upper yield point

C - Lower yield point

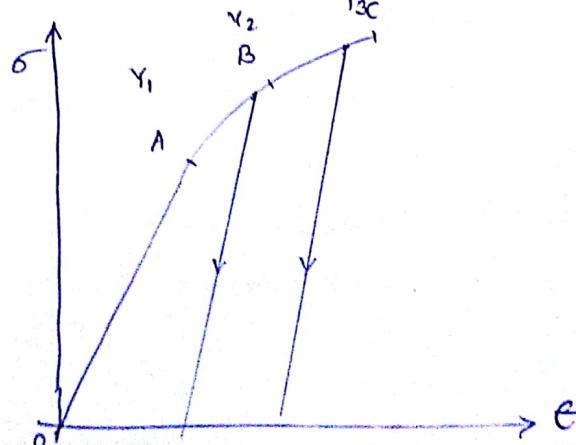
E - Ultimate tensile strength

* True stress-strain diag.: -

$$\epsilon = \int \frac{dl}{l} = \ln\left(\frac{l}{l_0}\right)$$



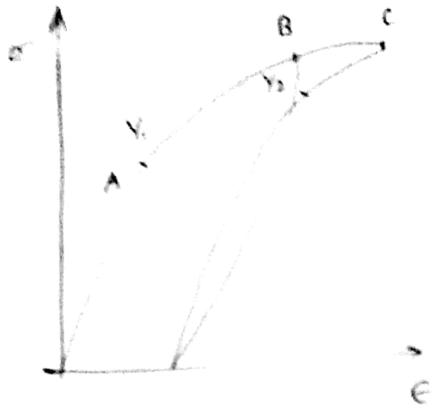
$$\epsilon_{12} = \ln \frac{l_1}{l_0} + \ln \frac{l_2}{l_1} = \ln \frac{l_2}{l_0}$$



OA : linear elastic

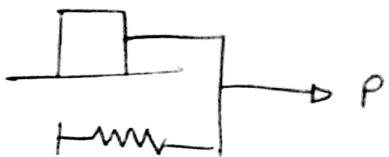
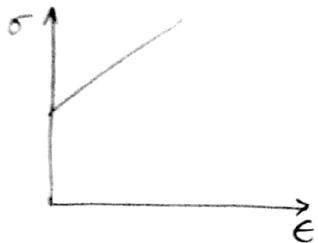
AB : uniform plastic def.

BC : Non-uniform plastic def.



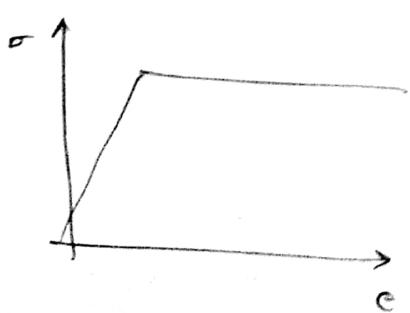
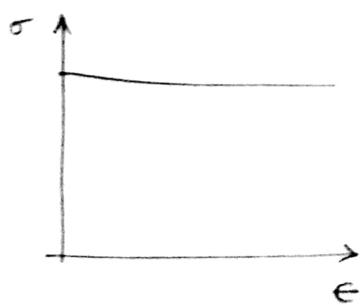
* Bauschinger effect:

Rigid + Linear work-hardening:



Total deformation = Elastic + Plastic

Materials \Rightarrow Rigid + Perfectly plastic
+ elastic



Casting

18/08/18

Quc. Following data were obtained from a plastic deformation part of load elongation test. The initial test width was 0.8 mm thickness / 12.5 mm width 50 mm gauge length.

Load (in kN)	Extension (mm)
1.57	0.08
1.90	0.76
2.24	1.85
2.57	3.66
2.78	5.84
2.90	8.92
2.93	11.06
2.94	13.49
2.92	16.59
2.86	19.48
2.61	21.82
2.18	22.69

- (a) Plot engg. stress vs engg. strain.
true stress vs true strain.

$$\log(\sigma) \text{ vs } \log(\epsilon)$$

- (b) Determine UTS and maximum load.

- (c) - h & n value of material

if it follows Holloman power hardening law

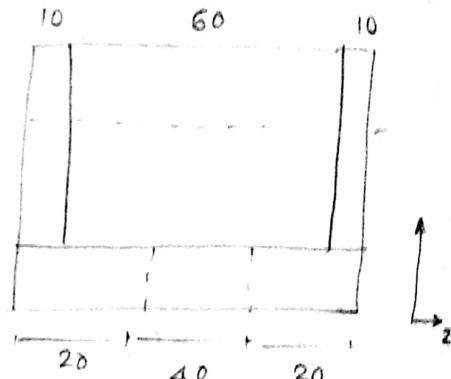
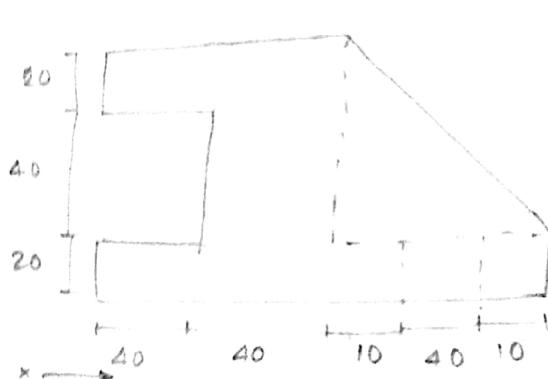
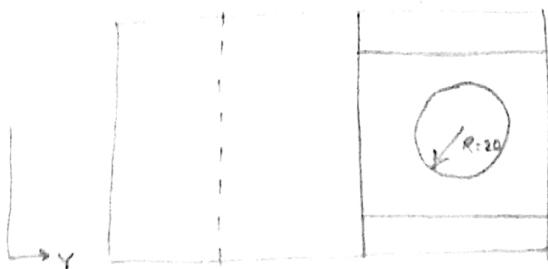
- Quc(D) Determine W.D if a bar of 10 mm dia. and 20 mm length is extended to 22 mm. The true stress-strain response of material is given as: $\sigma = 250 \epsilon^{0.3} \text{ N/mm}^2$

$$dw = \sigma A \cdot dl$$

$$= \sigma A l \frac{dl}{l} = \sigma A l (d\epsilon)$$

$$\therefore w = \int_0^{\epsilon} (Al) \cdot \sigma \cdot d\epsilon$$

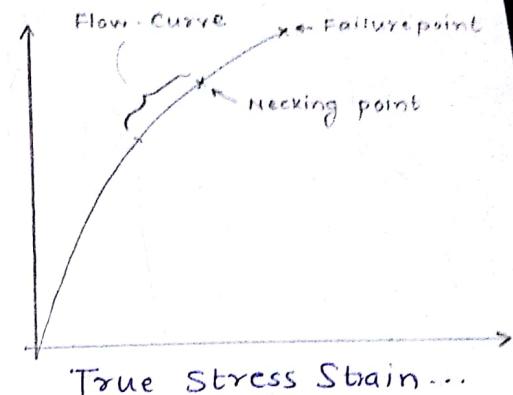
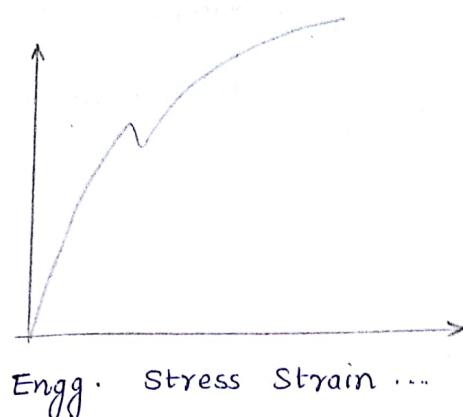
Que: For bracket casting shown, compare the 3 draw directions x, y, z (with parting line @ $x = 80\text{mm}$, $Y = 40\text{mm}$, $Z = 20\text{mm}$) in terms of draw distance & undercut volume.



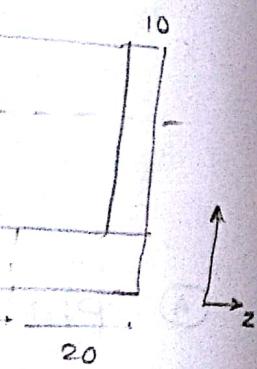
Parting pos.	No. of cores	Core volume	Max. draw distance
$X = 80\text{mm}$	1		80mm
$Y = 40\text{mm}$	1		40mm
$Z = 20\text{mm}$	2		60mm

Forming

14/08/2018



compare with
Parting
 $Z = 20 \text{ mm}$)
undercut



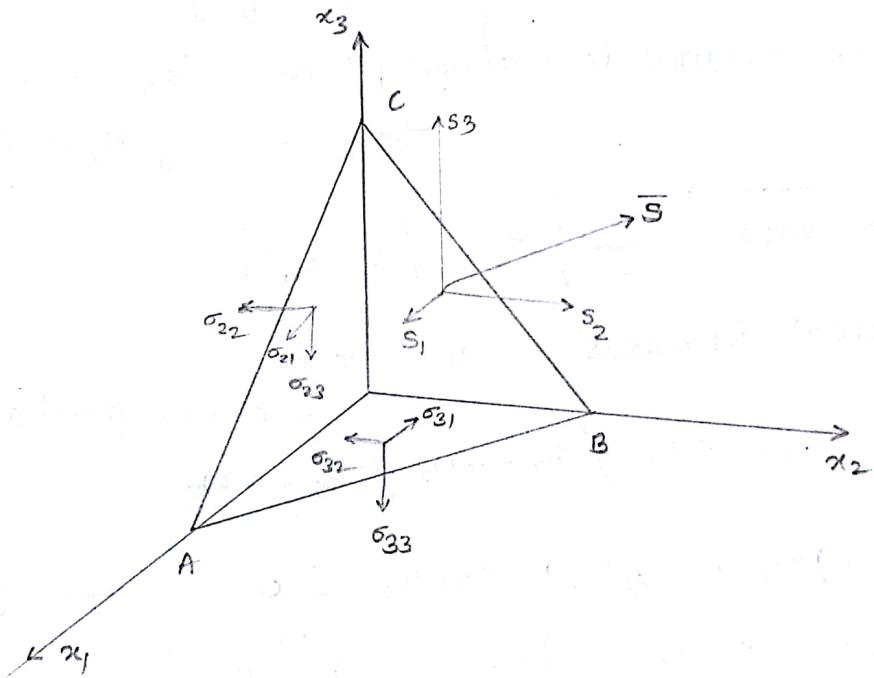
raw
stance

mm

mm

mm

- * Stress - Invariate: (Yield criteria)



$$\sum F_1 = \sum F_2 = \sum F_3 = 0 \quad [\sigma_{11}A_1 + \sigma_{22}A_2 + \sigma_{33}A_3 = 0]$$

- * Normal stress acting on plane ABC :

$$\sigma_n = S \cdot n$$

- * Principal stresses:

$$S_1 = \sigma n_1$$

$$S_2 = \sigma n_2$$

$$S_3 = \sigma n_3$$

$$S_1 A_s = \dots$$

$$S_2 A_s = \dots$$

$$S_3 A_s = \dots$$

Solve homogenous equation.

$$\# \text{ Total stress} = \text{Hydrostatic stress} + \text{Deviatoric stress}$$

↓

$$\frac{1}{3} (\sigma_{11} + \sigma_{22} + \sigma_{33})$$

↓
plastic shear

$$T = \begin{bmatrix} \sigma_{11} & \sigma_{12} & \sigma_{13} \\ \sigma_{21} & \sigma_{22} & \sigma_{23} \\ \sigma_{31} & \sigma_{32} & \sigma_{33} \end{bmatrix}$$

$$\boxed{\sigma'_{ij} = \sigma_{ij} + \frac{1}{3} \sigma_{xx} \delta_{ij}}$$

* From equil. stresses : $S_i = \sigma'_{ij} n_j$

* Stress normal to inclined plane : $\sigma_n = \bar{s} \cdot \bar{n}$
 $\Rightarrow \boxed{\sigma_n = \sigma'_{ij} n_j n_i}$

* Shear stresses components $\rightarrow \boxed{\sigma_s = \sqrt{s^2 - \sigma_n^2}}$

Principal Stresses : $\bar{s} = \sigma \bar{n}$
 ↴ scalar multiple.

$$S_1 = \sigma n_1 / S_2 = \sigma n_2 / S_3 = \sigma n_3$$

$$(\sigma_{11} - \sigma) n_1 + \sigma_{21} n_2 + \sigma_{31} n_3 = 0 \quad \text{--- (1)}$$

$$\sigma_{12} n_1 + (\sigma_{22} - \sigma) n_2 + \sigma_{32} n_3 = 0 \quad \text{--- (2)}$$

$$\sigma_{13} n_1 + \sigma_{23} n_2 + (\sigma_{33} - \sigma) = 0 \quad \text{--- (3)}$$

$$\det() = 0$$

$$\begin{vmatrix} \sigma_{11} - \sigma & \sigma_{21} & \sigma_{31} \\ \sigma_{12} & \sigma_{22} - \sigma & \sigma_{32} \\ \sigma_{13} & \sigma_{23} & \sigma_{33} - \sigma \end{vmatrix} = 0$$

Derivative
Deviatoric stress
↓
Plastic shear

$$\sigma^3 - I_1 \sigma^2 + I_2 \sigma - I_3 = 0$$

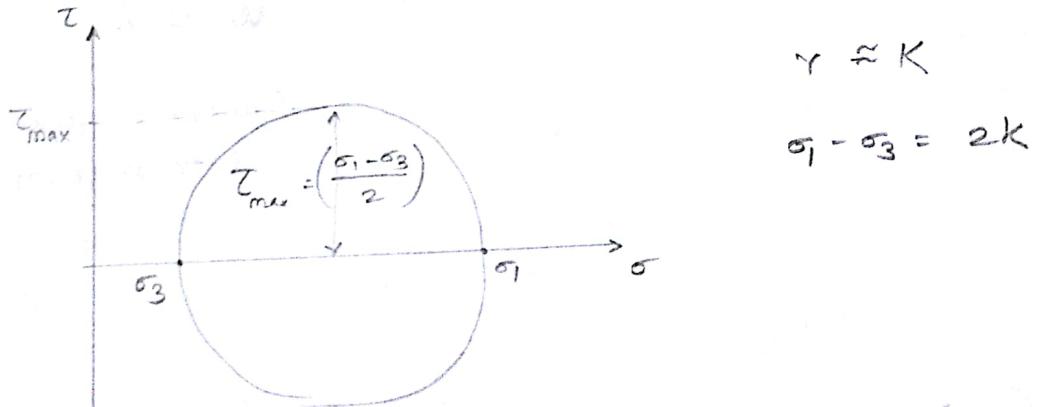
$$I_1 = \sigma_{11} + \sigma_{22} + \sigma_{33}$$

$$I_2 = \sigma_{11}\sigma_{22} + \sigma_{22}\sigma_{33} + \sigma_{33}\sigma_{11} - \sigma_{12}^2 - \sigma_{23}^2 - \sigma_{31}^2$$

$$I_3 = \sigma_{11}\sigma_{22}\sigma_{33} + 2\sigma_{12}\sigma_{23}\sigma_{31} - \sigma_{11}^2\sigma_{23}^2 - \sigma_{22}^2\sigma_{31}^2 - \sigma_{33}^2\sigma_{12}^2$$

From deviatoria : $\underbrace{J_1}_{\sigma_{ij}} \quad \underbrace{J_2}_{\sigma_{ij}} \quad \underbrace{J_3}_{\sigma_{ij}}$

$\Rightarrow \sigma > \gamma \rightarrow$ Yielding starts.
 Max. shear stress reaches a critical value
 Yielding Starts



$$\gamma \approx K$$

$$\sigma_1 - \sigma_3 = 2K$$

Shear strain energy per unit volume \rightarrow Reaches a critical value \rightarrow Yielding

$$(\sigma_1 - \sigma_2)^2 + (\sigma_2 - \sigma_3)^2 + (\sigma_3 - \sigma_1)^2 = \text{const} \equiv A$$

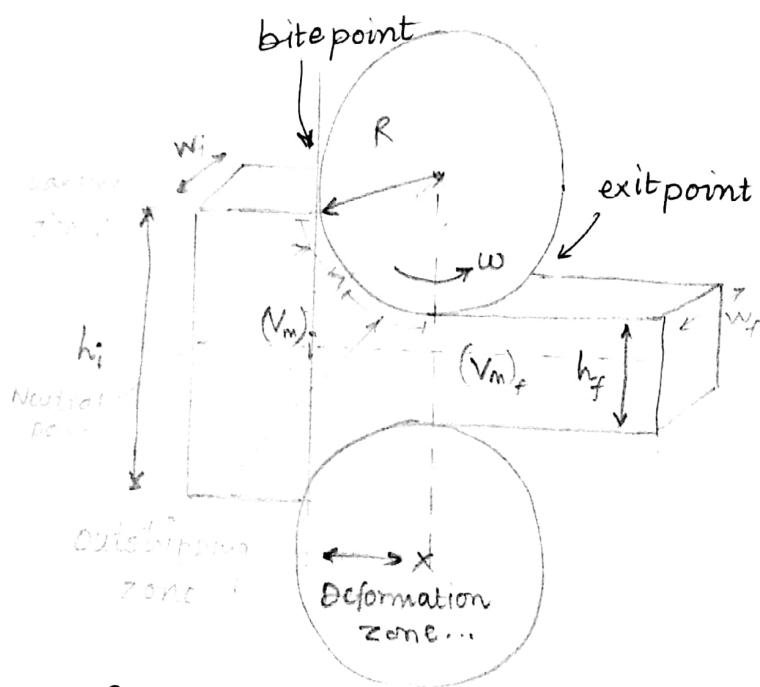
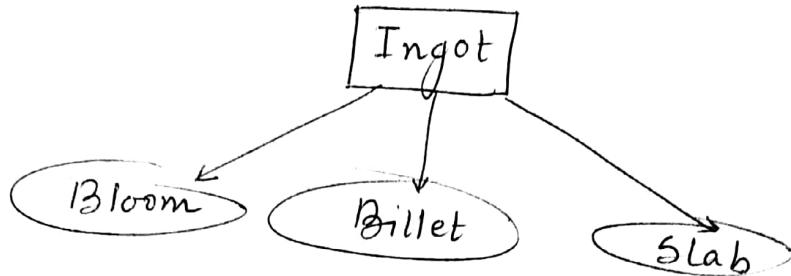
(1) $\sigma_1 = \gamma, \sigma_2 = 0 = \sigma_3$ \Rightarrow uniaxial loading.

(2) Pure torsion, $\sigma_1 = K, \sigma_2 = 0$

$$\sigma_3 = -K$$

* ROLLING : —

- Slab
 - Plate
 - Sheet
 - Strip
- ↓ below 6mm thickness



$$V = \omega * R$$

$$\omega \gg h$$

Plane-strain
deformation.

$$(V_m)_i < V_c (\equiv \omega R)$$

$$V = \text{const.}$$

$$w_i \sim w_f$$

$$l \cdot w \cdot h = \text{const}$$

$$d\epsilon_1 + d\epsilon_2 + d\epsilon_3 = 0$$

$$\therefore d\epsilon_e = -d\epsilon_h$$

Casting

* In an alloy, eutectic composition is such that liquidus temp. is minimum and freezing range is zero.

Q. Write briefly on:

- (1) Draft allowances
- (2) Machining allowances

(3) Distortion allowances (why residual stresses generate...)

Q. Brass chaplets are used to support sand core inside a sand mold cavity. The projected core print area is 13 cm^2 for each end of the cylindrical sand core which supports at both ends. The design of the chaplets and the manner in which they are placed in the mold cavity surface allows each chaplet to sustain a force of 45N. If vol. of the core = $7.5 \times 10^3 \text{ cm}^3$ and the metal poured is brass, determine the min. no. of chaplets that should be placed

(a) beneath the core.

(b) above the core

(Density of sand core & brass = 1.6 g/cm^3
green sand strength = $6.9 \times 10^3 \text{ N/m}^2$)

$$\rightarrow \text{Wt. of core} = 117.6 = W$$

$$\text{Support provided by core print at both ends} = 17.94 = S$$

$$\therefore \text{No. of chaplet at bottom} = \frac{W - S}{45}$$

* AFS grain fitness number: —

* Lifting properties of pattern: —

Q.1. A ladle with circular c/s (dia. = 2 m) contains 70×10^3 kg molten steel. The steel is teemed through circular hole (dia 3 cm) calculate time required to empty the ladle,

$$\rightarrow V_1 = \left(\frac{A_2}{A_1} \right) V_2 \\ = \frac{\pi (0.015)^2}{\pi (2/2)^2} \times V_2 = 2.25 \times 10^{-4} V_2$$

Neglecting V_1 compared to V_2 in Bernoulli,

$$V_2 = \sqrt{2gh} = \left(\frac{A_1}{A_2} \right) V_1$$

$$\sqrt{2gh} = \left(\frac{A_1}{A_2} \right) \frac{dh}{dt}$$

$$\therefore \frac{A_1}{A_2} \int_{h_0}^h \frac{-dh}{\sqrt{2gh}} = \int_0^t dt$$

$$\frac{-A_1}{A_2 \sqrt{2g}} \int_{h_0}^h \frac{dh}{\sqrt{h}} = t$$

* Let us assume that few atoms arrange themselves to a small nucleus/embryo of a spherical shape.

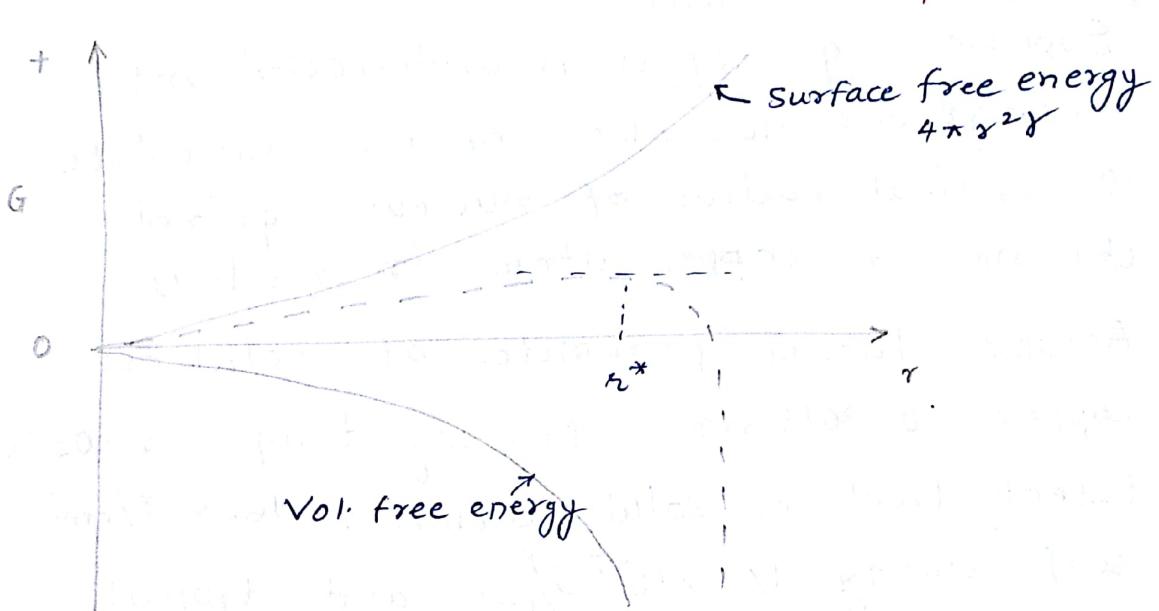
Let ΔG_v be the free energy per unit volume and γ is interfacial surface energy.

Nucleation of crystalline solid:

$$\text{Change in vol. free energy} = \frac{4}{3}\pi r^3 \Delta G_v$$

Energy req. to create new solid liquid interface = $4\pi r^2 \gamma$

$$\text{Net change in total free energy associated with formation of embryo} = \frac{4}{3}\pi r^3 \Delta G_v + 4\pi r^2 \gamma$$



There exists critical radius where barrier of solidification is maximum. (r^*)

$$\frac{d(\Delta G_T)}{dr} = 4\pi r^2 \Delta G_v + 8\pi r \gamma = 0$$

$$r = r^* = \frac{-2\gamma}{\Delta G_v}$$

$$\Delta G_T = \Delta G^* = \frac{16\pi r^3}{3 \Delta G_v^2}$$

There is a critical size of new phase (embryo) to become stable and grow. Hence there exists an energy barrier to solidification.

$$\Delta G_v = \frac{-2\gamma}{r^*} = -L \frac{\Delta T}{T_E}$$

$$\Delta T = T_E - T = \frac{2\gamma T_E}{L r^*}$$

$$T = T_E \left(1 - \frac{2\gamma}{L r^*} \right) \rightarrow \text{Gibbs Thompson Equation.}$$

Q.1 $T_1 < T_2 < T_E$

then draw $G_{T(\text{total})}$ w.r.t γ

- Q.2. Suppose liq. copper is undercooled until homogenous nucleation occurs. Calculate
 (a) critical radius of nucleus required,
 (b) no. of copper atoms in nucleus.

Assume lattice parameter of solid FCC copper 0.3615 nm , freezing temp. is 1085°C . Latent heat of solidification is 1628 J/cm^3 . surf. energy $177 \times 10^{-7} \text{ J/cm}^2$ and typical undercooling for homogenous nucleation is 236°C

$$\rightarrow \Delta T = 236^\circ\text{C}$$

$$T_E = 1085 + 273 = 1358$$

$$L = 1628 \text{ J/cm}^3$$

$$\gamma = 177 \times 10^{-7} \text{ J/cm}^2 \Rightarrow r^* = \frac{-2\gamma}{\Delta G_v} = \frac{-2 \times 177 \times 10^{-7}}{282.92} = -1.25 \times 10^{-7} \text{ cm}$$

$$\Delta G_v = \frac{-2\gamma}{r^*} = -L \frac{\Delta T}{T_E} = -1628 \times \frac{236}{1358} = 282.92 \text{ J/cm}^3$$

Phase
grow. Hence
to solidification

on Equation...

d until
calculate
required
nucleus.

1st Fcc

is 1085°C
 528 J/cm^3

typical
ion is 236°C

$$\frac{2 \times 177 \times 10^{-7}}{282.92} = 1.25 \times 10^{-7}$$

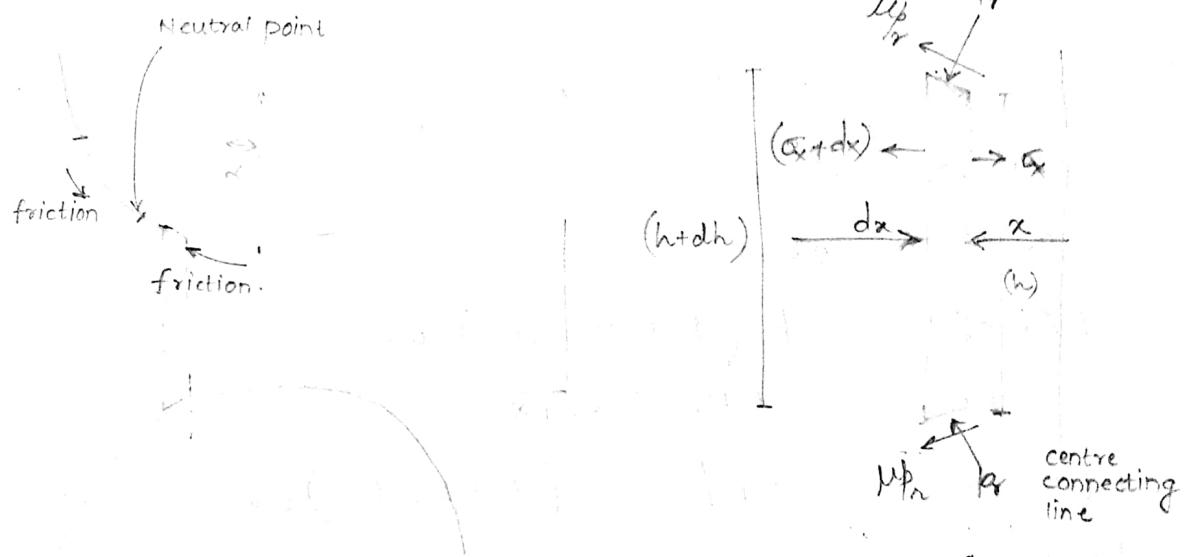
$$82.92 \text{ J/}$$

Forming

21/08/2018

1. Tresca Yield Criteria $\tau_{\max} = K$
2. von-mises " " $\left[\frac{\sigma_1 - \sigma_3}{2} \right] = K$
- Shear strain energy P_e
unit volume \rightarrow
 $(\sigma_1 - \sigma_3)^2 + (\sigma_2 - \sigma_3)^2 + (\sigma_3 - \sigma_1)^2 = A$
3. Uni-axial loading : $\sigma_1 = Y$
 $\sigma_2 = 0$
 $\sigma_3 = 0$

4. Pure torsion : $\sigma_1 = K$
 $\sigma_2 = 0$
 $\sigma_3 = -K$



- Due to longitudinal stress : $(\sigma_x + d\sigma_x)(h+dh)$
- Due to radial pressure : $2 p_r \left(\frac{dx}{\cos\alpha} \right) \sin\alpha$
- Due to frictional force : $2 \mu p_r \left(\frac{dx}{\cos\alpha} \right) \cos\alpha$

$$\sigma_1 = \sigma_x$$

$$\sigma_3 = \sigma_y = -p$$

$$* \sum F_x = 0$$

$$\Rightarrow h d\sigma_x + \sigma_x dh + 2 \int_r dx \tan \alpha + 2 \mu \int_r dx = 0$$

$$dh = 2 dx \tan \alpha$$

$$\rightarrow \sigma_1 = \sigma_x$$

$$\sigma_3 = \sigma_y = -\beta \quad [\text{vertical for.}]$$

* Normal to the direction of rotting \Rightarrow

$$\sigma_y dx = -\beta_r \frac{dx}{\cos \alpha} + \mu \beta_r \frac{dx}{\cos \alpha} \sin \alpha$$

$$\Rightarrow \sigma_y = -\beta_r (1 - \mu \tan \alpha)$$

$$\sigma_y = -\beta_r$$

$$\Rightarrow \sigma_3 = \sigma_1 = -\beta = -\beta_r$$

Tresca Yield Criteria:

$$\boxed{\sigma_x + \beta = 2k}$$

$$d(h\sigma_x) = -\beta (1 \pm \mu \cot \alpha) dh$$

$$dh = 2(R d\alpha) \sin \alpha$$

$$d(h_s - h_p) = -2R\beta \sin \alpha (1 \pm \mu \cot \alpha) d\alpha$$

$$\frac{d}{d\alpha} \left[h_s \left(1 - \frac{\beta}{s} \right) \right] = -2R\beta \sin \alpha (1 \pm \mu \cot \alpha)$$

$$\Rightarrow h_s \frac{d}{d\alpha} \left(1 - \frac{\beta}{s} \right) = -2R\beta (\sin \alpha \pm \mu \cos \alpha)$$

$$h = h_s + (2R) \frac{\alpha^2}{2}$$

$$\Rightarrow \frac{d(P/s)}{P/s} = \frac{2\alpha d\alpha}{(\frac{h_s}{R} + \alpha^2)} \pm \frac{2\mu d\alpha}{(\frac{h_s}{R} + \alpha^2)}$$

$$\Rightarrow \ln\left(\frac{P}{S}\right) = \ln\left(\frac{h_a}{R} + \alpha^2\right) \pm 2\mu \frac{1}{\sqrt{\frac{h_a}{R}}} \tan^{-1}\left(\frac{\alpha}{\sqrt{\frac{h_a}{R}}}\right)$$

$$\Rightarrow \boxed{\frac{P^+}{S} = C^+\left(\frac{h}{R}\right) e^{\mu H}} \quad \left. \begin{array}{l} \\ \end{array} \right\} \text{on exit side}$$

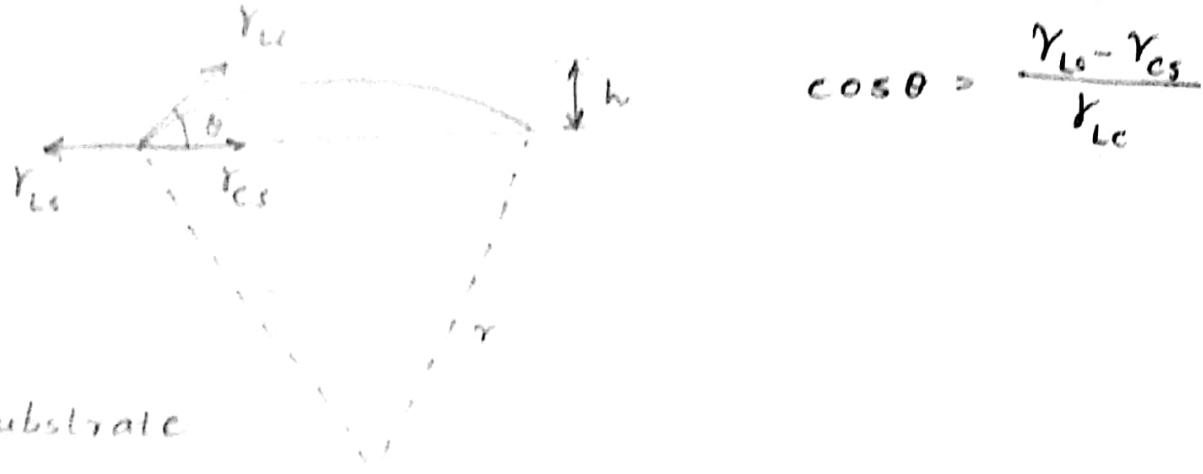
$$\Rightarrow \boxed{\frac{P^-}{S} = \bar{C}\left(\frac{h}{R}\right) e^{-\mu H}} \quad \left. \begin{array}{l} \\ \end{array} \right\} \text{on entry side}$$

$$\Rightarrow \boxed{H = 2 \sqrt{\frac{R}{h_a}} \tan^{-1}\left(\sqrt{\frac{R}{h_a}} \alpha\right)}$$

Casting

21/08/2019

- * (problem solving)
- * Nucleation due to nucleating agents:
Liquid



$$\cos \theta = \frac{\gamma_{LG} - \gamma_{CS}}{\gamma_{LG}}$$

substrate

$$\therefore V = \frac{1}{3} \pi h^2 (3r - h)$$

$$= \frac{1}{3} \pi h^3 (2 - 3 \cos \theta + \cos^3 \theta)$$

$$\therefore A = 2\pi rh = 2\pi r^2 (1 - \cos \theta)$$

$$\therefore \Delta G_T = \gamma_{LG} \cdot 2\pi r^2 (1 - \cos \theta) + \frac{1}{3} \pi r^3 (2 - 3 \cos \theta + \cos^3 \theta) \Delta G$$

$$+ (\gamma_{CS} - \gamma_{SL}) \pi r^2 (1 - \cos^2 \theta)$$

$$\therefore \gamma^* = \frac{2\gamma_{LG}}{\Delta G_T}$$

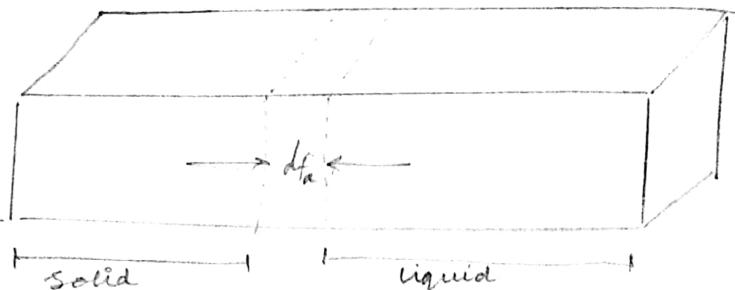
$$\therefore \Delta G^* = 4\pi \gamma_{LG}^3 (2 - 3 \cos \theta + \cos^3 \theta)$$

Segregation during solidification:

$$k_o = \frac{C_s}{C_L} \rightarrow \text{partition coeff.}$$

composition of solid / liq.

- composition of solid @ advancing solid-liq. interface -



A liq. metal of initial composition C_0 is solidifying within the rectangle.

Solidification point is propagating from left to right

Let f_s be fraction solidified.

Now, composition of liq. is C_L

At any small instant of time, volume fraction of df_s suppose solidifies.

Let conc. of the small elemental front is C_s^*

$$(C_L - C_s^*) df_s = (1 - f_s) dC_L$$

$$\therefore (C_L - k_o C_L) df_s = (1 - f_s) dC_L$$

$$\int_0^{f_s} \frac{df_s}{1 - f_s} = \int_{C_0}^{C_L} \frac{dC_L}{C_L (1 - k_o)}$$

$$C_L = C_0 (1 - f_s)^{k_o - 1}$$

$$C_s^* = k_o C_L = k_o C_0 (1 - f_s)^{k_o - 1}$$

Jcheil Equation...

Welding

23/08/2017

* Material Transfer types -

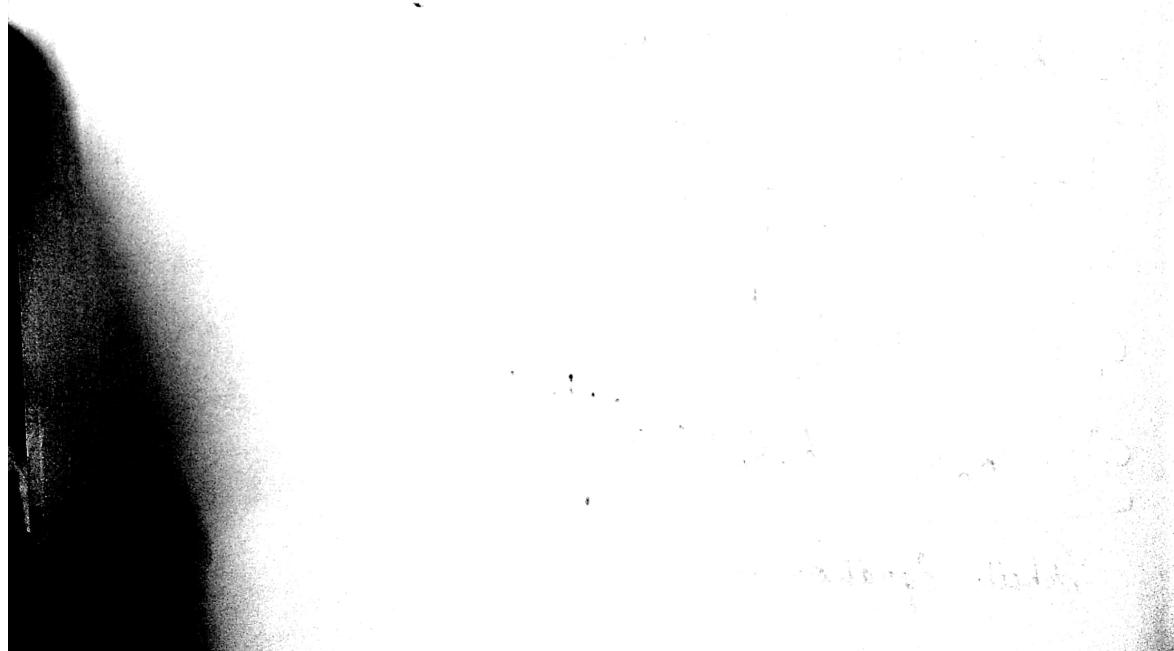
- ① Free flight · Globular - projected transfer:
 - Effect of increasing current.
- ② Bridging or short circuit transfer:
 - Sequence of above process



③ Pulsed arc or pulsed current transfer;

* Difference b/w normal spray mode and pulsed transfer type of metal transfer -

④ Slag protected metal transfer;



* Pressure distribution during rolling:-
[discussed in previous class]

• Boundary conditions:

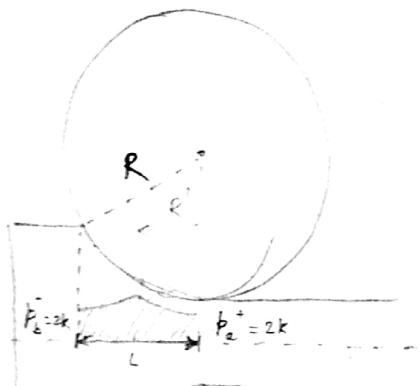
① No pull at front and back ends.

$$\text{i.e. } \sigma_{xa} = 0 \rightarrow H_a = 0 \text{ (at exit point)}$$

$$\text{Exit point} \Rightarrow p_a^+ = 2k \Rightarrow c^+ = \frac{R}{h_a}$$

$$\text{Bite point} \Rightarrow p_b^- = 2k \quad * \left(\frac{p^+}{s} \right) = \left(\frac{h}{h_a} \right) e^{uH}$$

$$* \left(\frac{p^-}{s} \right) = \left(\frac{h}{h_a} \right) e^{u(H_0 - H)}$$



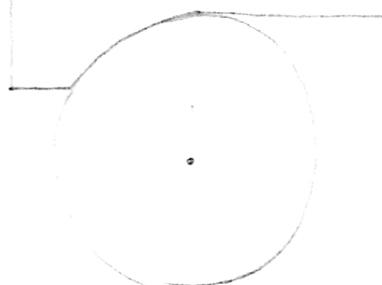
Front pull $\Rightarrow F_a \checkmark$

$$\underbrace{\sigma_{xa}}_{\sigma_x + p = 2k} \checkmark$$

$$p = 2k - \sigma_{xa}$$

$$(p_a^+) < (p_a^+)$$

With pull No front pull



$$\sigma_{xb} + p^- = 2k$$

$$(p_b^-) < (p_b^-)$$

No back
pull

with
pull

② Roller radius: $R' < R$ preferable as it decreases the 'L'

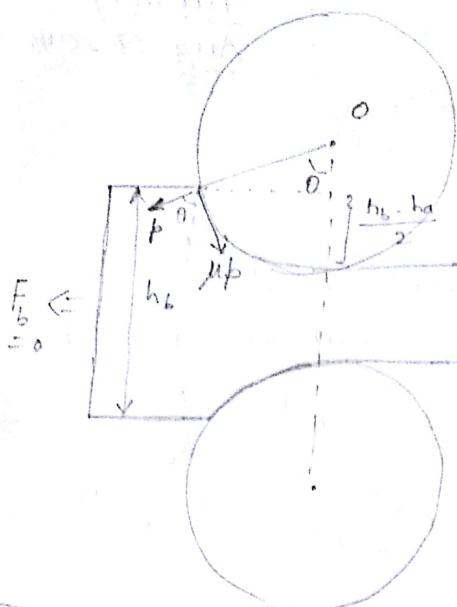
③ Rolling Load : $F = \int_{\alpha_n}^{\theta} p_r() + \int_0^{\alpha_n} p_s()$

$$\text{Torque} = \int \mu p_r() + \int \mu p_s()$$

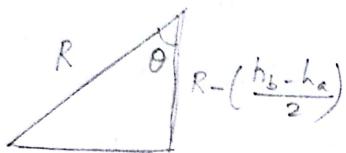
$$\text{Power} = T \cdot w$$

$$\mu \geq \tan \theta$$

$$p \sin \theta < \mu p \cos \theta$$



Draft for unaided rolling ...



$$\cos \theta = \frac{R - \frac{\Delta h}{2}}{R}$$

$$\Rightarrow \cos \theta = 1 - \frac{\Delta h}{2R}$$

$$\therefore \frac{\Delta h}{2R} = 1 - \cos \theta$$

$$\therefore \frac{\sqrt{R^2 - (R - \frac{\Delta h}{2})^2}}{R} = \tan \theta$$

$$\Rightarrow \sqrt{\frac{\Delta h}{R}} = \mu \rightarrow \Delta h = \mu^2 R$$

Rise

- C
- C
- V

Feed

- L

① U

②

[#]

③

④

⑤

⑥

⑦

Casting...

Tuesday
Aug. 28. 2018

Riser :-

- Completes the mould filling.
- Compromises liquid and solidification shrinkage.
- Venting entrapped gases.

Feeder design :

- Location, type and dimensions of feeder need to be defined / decided.
- 7 feeding rules :

① Unnecessary feeder decreases casting yield.

② Time of solidification of riser should be higher than that of the casting.

[# Caine's method : Vol. ratio vs. freezing ratio]

③ Feeder should contain sufficient liquid metal to feed the contraction requirement of casting and itself.

④ There must be a path to reach the feed metal to req. region. This is achieved if there is directional solidification of casting as a whole towards riser.

[$M_r \sim 1.2 \text{ or } 1.1 M_c$: Modulus of riser/casting]

⑤ Junction requirement : [side riser]

⑥ Pressure requirement :

⑦ Temperature gradient :

$$* \text{ No. of risers} = \frac{100 - 2EE}{2RE + d}$$

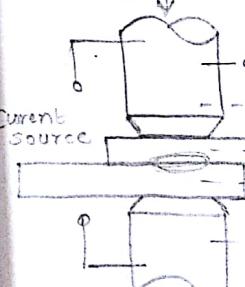
Riser: $d = 4.95''$	$RE = 1.5t$	$t = \text{some time after pouring.}$
$h = 9.9''$	$EE = 2t$	

Location of hotspot Using MATLAB :

- ① Divide the body into several section
- ② Imagine a point within the section. Join the end points to the point in order to discretize the section.
- ③ Evaluate the volume and surface area of each discretized element.
- ④ Draw the vector whose magnitude is $\sqrt{V/A}$ and direction is bisecting internal angle made @ the imaginary point.
- ⑤ Addition of vector of all discretized element
- ⑥ Resultant vector should be the next imaginary point.
- ⑦ Loop continues until we reach hotspot area.

- * Arc Welding
 - Constant
 - Constant
 - . Self Reheat rate.
- Combined
- * Thermite
 - Metal Oxide
 - Heat for superheating
- Ex: 2A

* Resistance



$$Q = Pt$$

$$\Delta T = \frac{Q}{mC}$$

- Copper
- No. of layers
- Solid

Welding

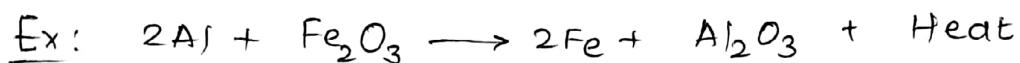
Thursday
30-08-2018

* Arc Welding :

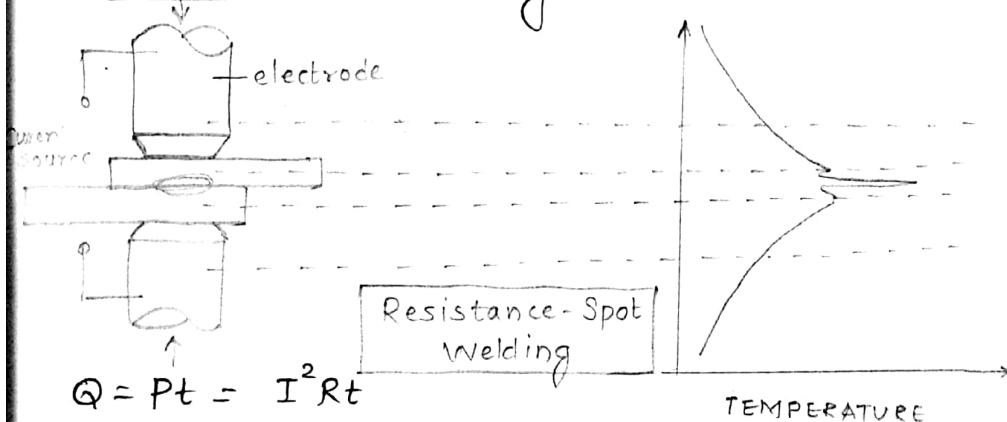
- Constant current power source.
- Constant voltage power source.
 - 'Self Regulation' - Tries to maintain arc length by changing the metal consumption rate.
- Combined characteristic sources.

* Thermite Mixture :

- Metal oxide + Aluminium \rightarrow Metal + Al oxide + heat
- Heat for coalescence is produced by the superheated molten metal.



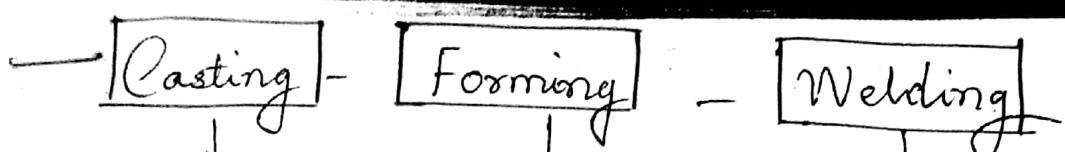
* Resistance Welding :



$$Q = Pt = I^2 R t$$

$$\Delta T = \frac{Q}{mc_p} = \frac{I^2 R t}{mc_p}$$

- copper electrode \rightarrow Highly conductive : There is no melting @ electrode-plate junction.
- Solid state \leftrightarrow Fusion { combination of both }



Prof. S.K. Panda
(Monday)

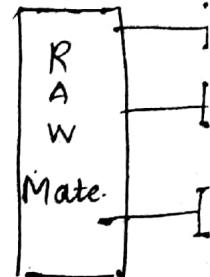
S. K. Pal
(Tuesday)

J. Paul.
(Thursday)

7 miles
↓
STC (Foundry)

7 miles
↓
Inside STC

6 miles → TA
↓
ME dpt. ground floor
+
Friction stir welding
(STC)



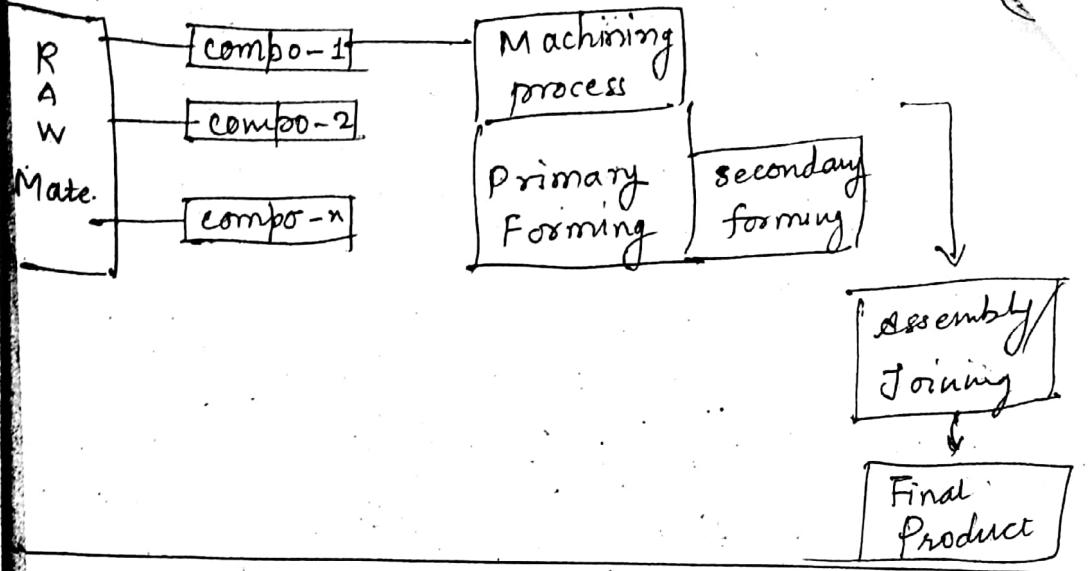
* Syllabus [Forming] :

1. Crystal Structure
2. Phase Diagram
3. T-T-T diagram
4. Heat treatment process
5. Deformation Processes.
6. Yield criteria [Tresca + von-mises yield criteria]
7. Bulk metal forming process.
 - ① Rolling
 - ② Drawing → Rod / Bar
→ Tube drawing
 - ③ Extrusion
 - ④ Forging
8. Sheet metal forming [Deep drawing process]
9. Non-conventional forming
 - Electromagn.
 - Explosive
 - Electrohydraulic.
10. Powder metallurgy
11. Mise process

- ① FSW Lab (Dept.)
- ② CoE in Adv. M.Tech.
- ③ HMC (13 Sept. 2018)

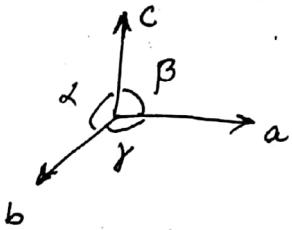
- * CRYSTAL
- ① 3D s
- ② some
- ③ (1) Triclinic
(2) Mono-c
(3) Ortho.
(4) Rhom.
(5) Tetra.
(6) Hexag.
(7) cubic
- ④ Depend.
- ⑤ cubic





CRYSTAL STRUCTURE : -

-) 3D structure - space lattice
-) smallest unit — unit cell.



a, b, c
 α, β, γ

- (1) Triclinic
- (2) Mono-clinic
- (3) Orthorhombic
- (4) Rhombohedral
- (5) Tetragonal
- (6) Hexagonal
- (7) Cubic

Crystal system

-) Depends on: no. of atoms present in a particular crystal sys.

Cubic - 1) [Simple cubic [SC]]

$$8 \text{ corners } \left[\frac{1}{8} \times 8 = 1 \right] = E \cdot N \cdot A$$
 (effective no. of atom)

Packing efficiency = $\frac{\text{Vol. of atoms}}{\text{Total Vol.}}$

$$P.F. = \frac{\frac{4}{3} \pi r_a^3}{a^3}$$

$$a = 2r_a$$

2) Body centred cubic:

$$E.N.A = 1 + \left(\frac{1}{8} \times 8 \right) = 2$$

$$P.F = \frac{2 \times \frac{4}{3} \pi r_a^3}{a^2}$$

$$\sqrt{3}a = 4r$$

3) Face centred cubic:

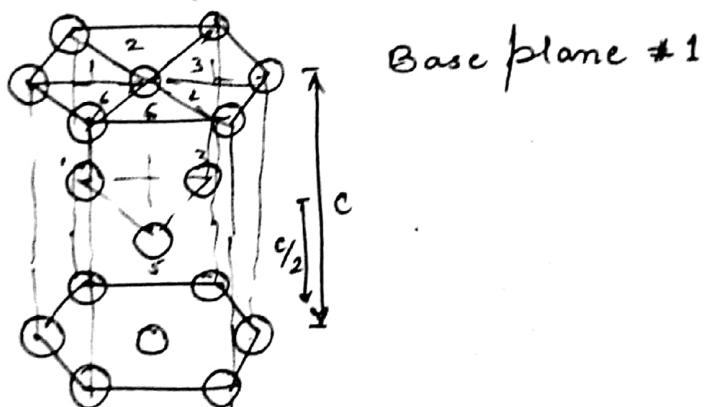
$$E.N.A = \left(6 \times \frac{1}{2} \right) + \left(8 \times \frac{1}{8} \right) = 4$$

$$P.F = \frac{4 \times \frac{4}{3} \pi r_a^3}{a^2}$$

$$\sqrt{2}a = 4r$$

⑥ Hexagonal -

$a = b \neq c$ (HCP-hexagonal closed packed)



Top



$$E.N.A = 3 + \left(\frac{1}{6} \times 12 \right) + \left(\frac{1}{2} \times 2 \right)$$

$$= 6$$

$$VOL \text{ of unit cell} = (6 \times eq \cdot \Delta^6) \times c$$

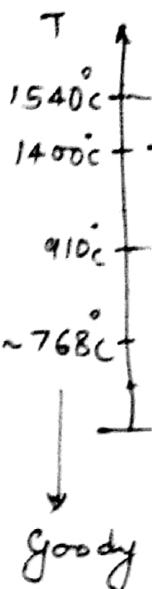
Rel. b/w r_a and c

$$P.F = \frac{6 \times \frac{4}{3} \pi r_a^3}{6 \times eq \Delta^6 \times c}$$

(3) Al/
Fe -

Ti

(8) Iron



(9) Allo.



Carbo

(10) Dope

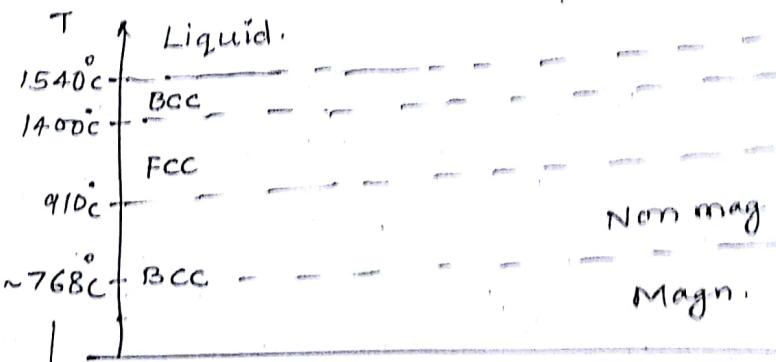
(1) Ti

(2) v

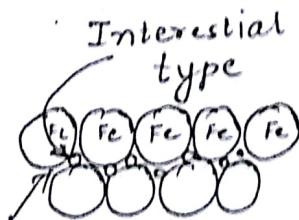
- Pha

(3)

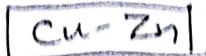
(7) $\text{Al}/\text{Cu} \rightarrow \text{FCC}$
 $\text{Fe} \xrightarrow{\text{FCC}} \text{BCC}$ { Depends on temp.
 (austenitic)}

 $\text{Ti} \rightarrow \text{HCP}$ (8) Iron \rightarrow Melt. Ph. $\rightarrow 1540^\circ\text{C}$ 

Goodey point - Transition ph. b/w magnetic
and non-magnetic.

(9) Alloy - Solid Solutions

Substitution type



Carbon

(10) Dependency of crystal structure on -

(1) Temp.

(2) % of solute in the alloy.

- Phase equil. diag.

Welding

19/07/2018

1. Principles of Welding, Robert Messler

(5)

2. Joining → Welding

* Condu

Brazing

* Conve

Soldering

* Radiat

Adhesive Bonding

* Therm

3. 2 thin sheets → Threaded fasteners, etc.

Bottom plate

.. R_{th}

is thick → Screw

: R_{th}

4. Localised coalescence achieved by appropriate combination of temp. & pressure.

: R_{th}

5. Fusion Welding - Melting the base material.

* Ste

6. Solid State welding - The temp. is maintained below the m.p. of the base material.

• Temp

7. Types of Joint -

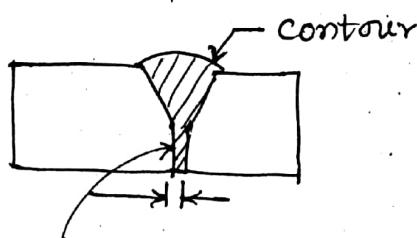
• Hea

& Fillet Weld ∇ Groove Weld

* Bot

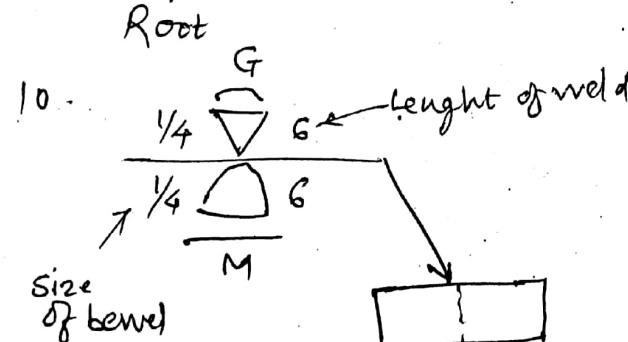
8. Weld Symbols →

(1) Fu



9. Weld Symbols →

(2) -



(3) J.

(4) S.

* 1-T

11.

H

Δx

12.

H

Δx

13.

H

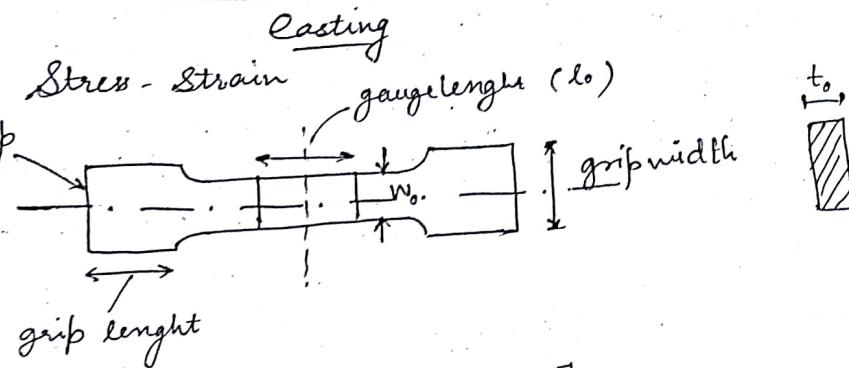
Δx

14.

H

Δx

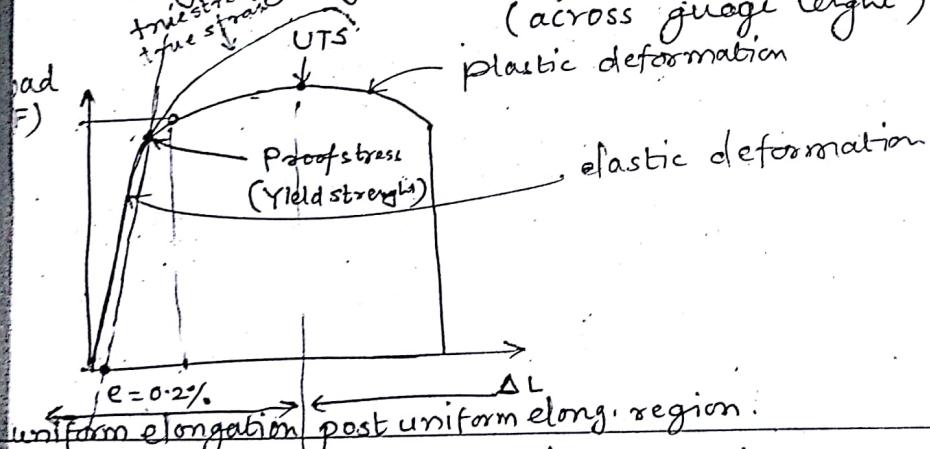
(6)



[ASTM, tensile specimen]

sensors:

- Load sensor (Load cell) - Load on specimen.
- Strength guage - displacement in specimen
(across gauge length)

Initial area of c/s = $A_0 = w_0 \times t_0$

$$\text{Engg. stress (S)} = \frac{F}{A_0}$$

$$\text{Engg. strain (e)} = \frac{\Delta l}{l_0}$$

Hooke's Law: $\sigma = E \epsilon$ * Yield str: @ $e = 0.2\%$ draw line || to graph.

* UTS: Ultimate Tensile Strength.

① During Tensile Test 2 phenomena take place

① Decr. in load bearing cap. of specimen.
due to decr. in area of c/s

② [Vol. remains const.]

② Load bearing capacity of specimen incr.
due to strain hardening.

* UTS - instability point
(Onset of necking)

* True stress and true strain:-
↓

$$\sigma = \frac{F}{A}$$

$A \rightarrow$ instantaneous area of c/s $\Rightarrow \sigma = \frac{F}{A_0} \times \frac{A_0}{A}$

$$\text{As, } A_0 l_0 = A l \quad (\text{vol.})$$

$$\therefore \frac{A_0}{A} = \frac{l}{l_0}$$

$$\therefore \sigma = \left(\frac{F}{A_0} \right) \left(\frac{l}{l_0} \right) = \underset{\substack{\text{engg} \\ \text{stress}}}{\downarrow} s \underset{\substack{\text{engg} \\ \text{strain}}}{\downarrow} (1+e)$$

* True strain:

$l \rightarrow$ instantaneous length

$dl \rightarrow$ change in length

$$d\epsilon = \frac{dl}{l}$$

$$\therefore \epsilon = \int_{l_0}^l \frac{dl}{l} = \ln \left(\frac{l}{l_0} \right) = \ln (1+e)$$

Q1 A bar of length l_0 is uniformly extended until its length $l = 2l_0$. (3)

① Compute engg. strain and true strain

② To what final length must a bar of length l_0 be compressed if strains are to be same except the sign in part A.

$$\rightarrow \text{① Engg. strain} = 1 = e$$

$$\text{True strain} = \ln 2 = 0.693$$

② $e = -1$, not possible. \rightarrow engg. stra

$$e = -0.693 \rightarrow 1 - 0.693 = \frac{l}{l_0}$$

$$\Rightarrow l = 0.$$

True strain

For equivalent amt. of tension $\rightarrow 0.693$

compression $\rightarrow -0.693$

Tensile strain for eq. amt. of tension and compression are same except sign.

Q2 A bar of 10cm elongated to 20cm in 3 steps: 10cm \rightarrow 12cm \rightarrow 15cm \rightarrow 20cm

① calc. engg. strain for each step, compare sum with overall strain.

② Repeat for true strains:

$$\rightarrow 1) e_1 = \frac{2}{10} = 0.2 \quad e = 1$$

$$e_2 = \frac{3}{12} = 0.25$$

$$e_3 = \frac{5}{15} = 0.33$$

$$2) \bar{e}_1 = \ln(1.2) \quad \bar{e} = \ln(2) = 0.693$$

$$\bar{e}_2 = \ln(1.25)$$

$$\bar{e}_3 = \ln(1.33)$$

True strains are additive in nature.

(1+e)

$$* l_0 \times w_0 \times t_0 = l \times w \times t$$

$$\frac{l_0}{l} \times \frac{w}{w_0} \times \frac{t}{t_0} = 1$$

$$\therefore \ln\left(\frac{l}{l_0}\right) + \ln\left(\frac{w}{w_0}\right) + \ln\left(\frac{t}{t_0}\right) = 0$$

$$\therefore \epsilon_x + \epsilon_w + \epsilon_t = 0$$

* W.O. constancy cond. can be expressed using simple mathematical cond. in terms of true strain.

Hollomon Power hardening Law:-

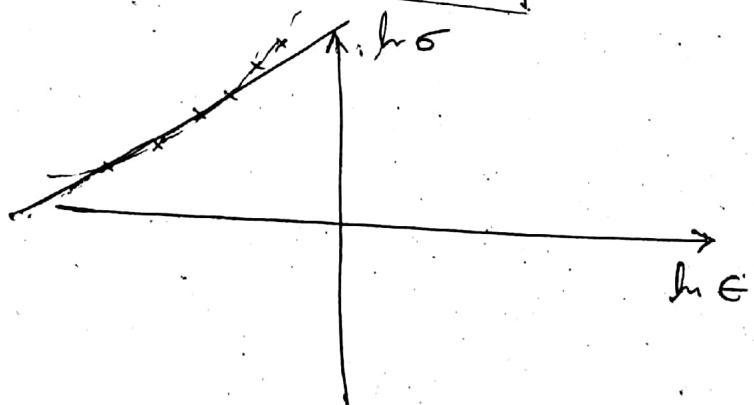
$$\boxed{\sigma = K \epsilon^n}$$

~~During post elongation regime~~

K = Strength coefficient.

n = strain hardening exponent....

$$\ln \sigma = n \ln \epsilon + \ln K$$



Yield Strength — UTS (applicability)

Profit:

Psycho well
the func
to

4. We
for
5. Mar

- P
I
-

Metal Forming

24/07/2018

(10)

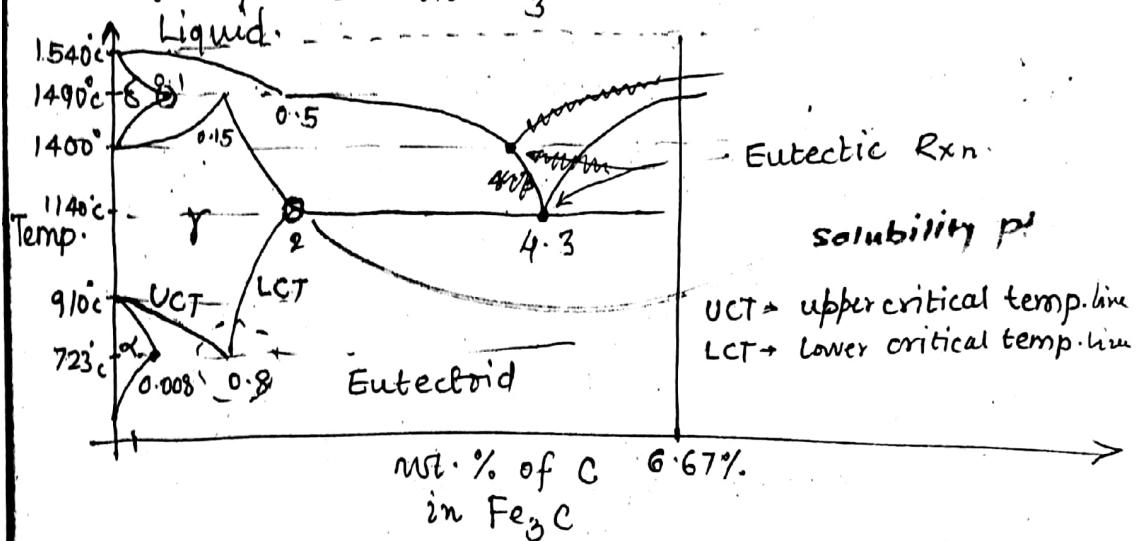
Crystal Structure of Metal :-

$$[\text{Alloy of Fe}] = \text{Fe} + \text{C/N}/\dots$$

Solution Solvent Solute

- solid solution.
- solubility of C/N in Fe lattice depends on the temperature.

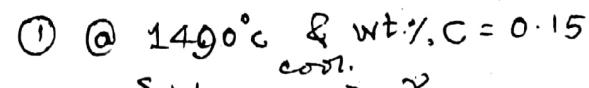
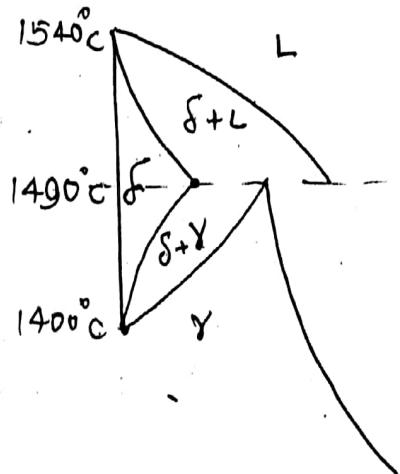
wt. % of C in Fe_3C : (Iron Carbide)



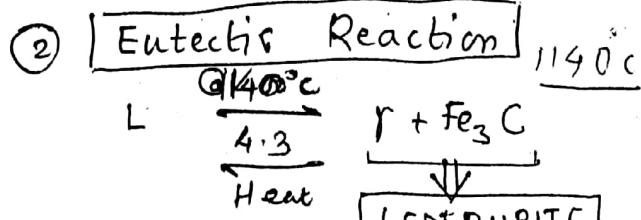
Phase diag. of Fe and Fe_3C .

- Cementite - Orthorhombic structure (6.67% C)
 - Max. solubility of C.

- α, δ - BCC
- γ - FCC

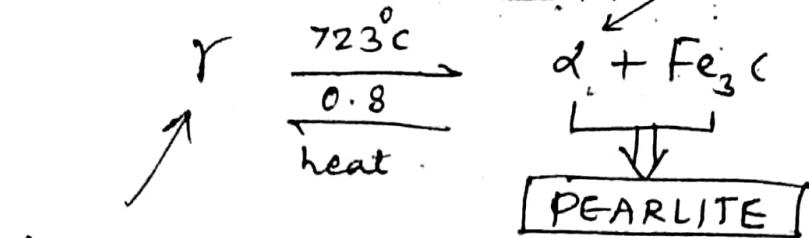


Peritectic Reaction



Ferrite (softest) ①

③ Eutectoid Reaction



Austenite

* Important metal forming reaction.

[Tensile Strength / Yield Strength

$$T_S / Y_S = f(\alpha, \beta, \gamma, f, d)$$

1. Cast Iron - Brittle (more amt of C):

above 2% wt.

2. Steel - below 2% wt. C.

3. Hypo-eutectic cast iron - bet" 2% - 4.3%

4. Hyper " " " > 4.3%.

5. Hypo-eutectoid steel < 0.8% C

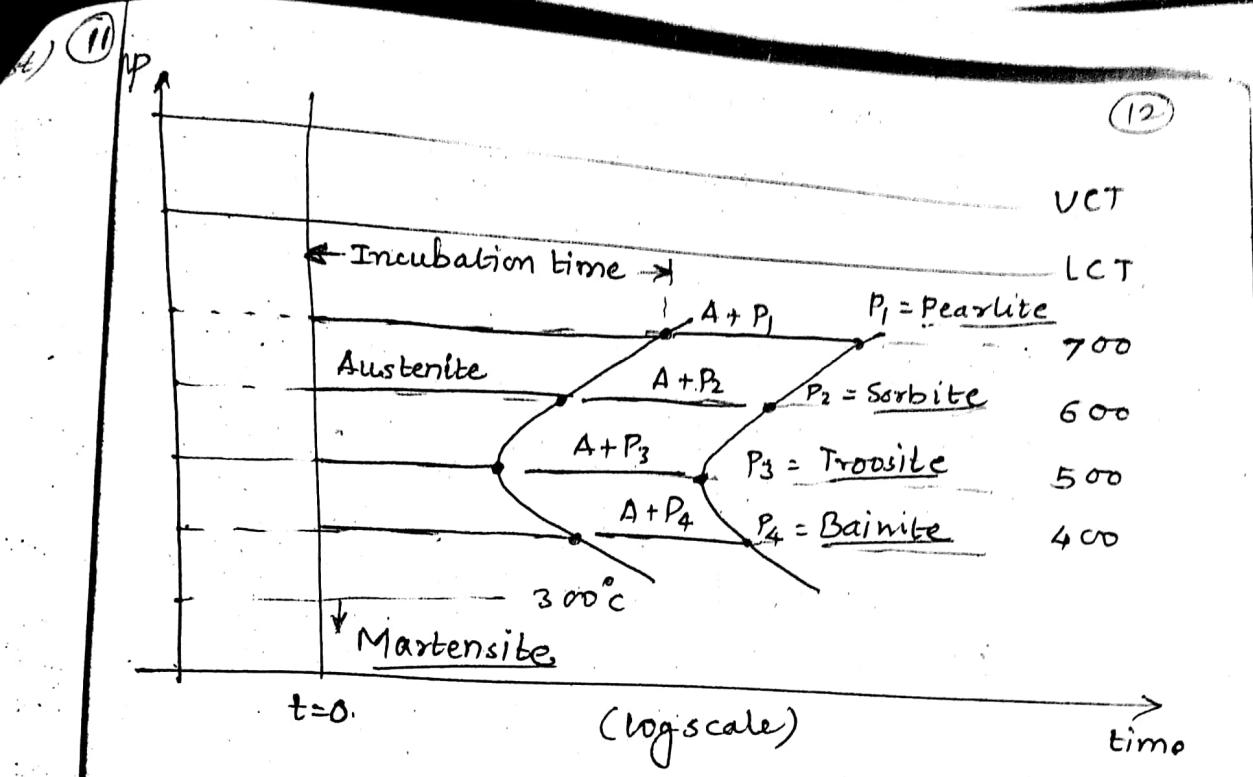
6. Hyper - " " : 0.8% - 2%

7. Low carbon steel = upto 0.3%

8. Medium carbon " = 0.3 → 0.6%

9. High carbon " = 0.6 → 1%

10. Tool steel = 1 → 2%



Diffusional transformation: $f(\text{temp., time})$

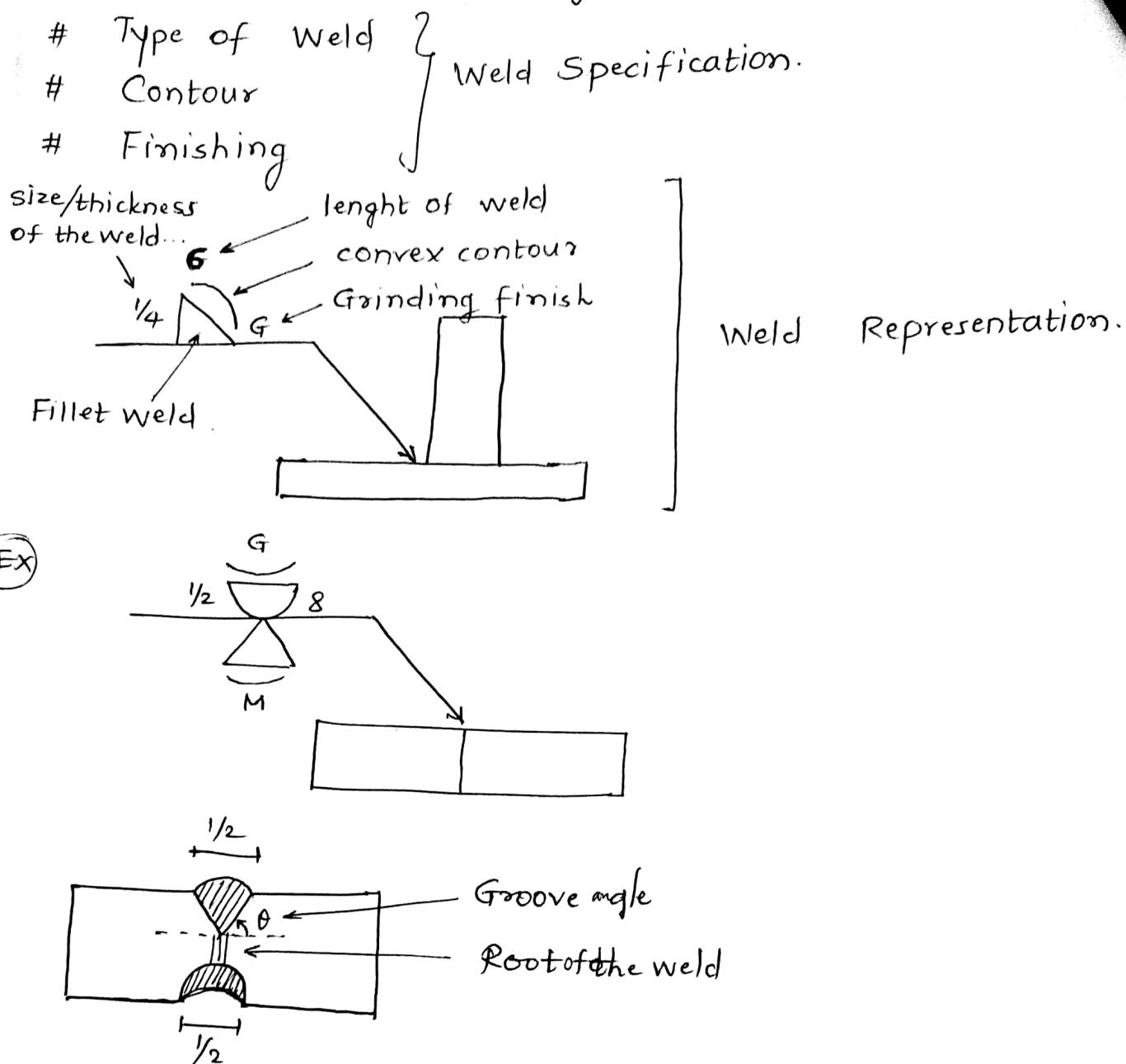
Austenite \longleftrightarrow Martensite : $f(\text{temp.})$

[Non-diffused : A doesn't fully transform to M.]

$$\begin{array}{l} TS \\ \searrow \\ \gamma S \end{array} \rightarrow f(\alpha, \beta, r, \delta, d)$$

26/07/2018

Welding



① Elements of welding Setup -

- Energy Source
- Removal of surface contaminants.
- Atmospheric contaminants.
- Weld Metallurgy

② Fusion Welding - Temp. above m.p.t.

③ Solid-state welding - Temp. below m.p.t.

Casting

30/07/2018

(14)

n-value :-

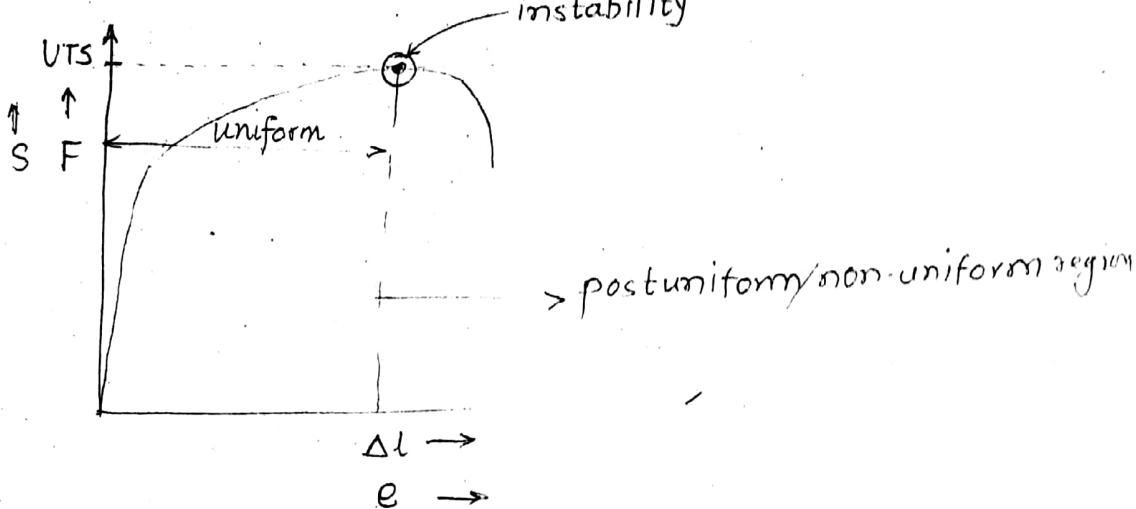
$$\sigma = K \epsilon^n \quad |$$

- Physical significance: - strain coeff.

$$F = \sigma A$$

$$dF = \sigma dA + Ad\sigma$$

$$\therefore \frac{dF}{F} = \frac{dA}{A} + \frac{d\sigma}{\sigma} \quad | \quad \text{---} \quad (1)$$



$$\text{At UTS, } (1): \frac{d\sigma}{\sigma} + \frac{dA}{A} = 0 \quad | \quad (2)$$

$$\frac{d\sigma}{\sigma} = -\frac{dA}{A} \quad | \quad (3)$$

As, V is constant

$$\frac{dV}{V} = \frac{dA}{A} + \frac{dl}{l}$$

$$\therefore \frac{dA}{A} = -\frac{dl}{l} \quad | \quad (4)$$

Substituting in (3),

$$\frac{d\sigma}{\sigma} = \frac{dl}{l} = d\epsilon$$

$$\frac{d\sigma}{d\epsilon} = \sigma \quad | \quad \text{at UTS}$$

$$\therefore \frac{d(K\epsilon^n)}{d\epsilon} \Big|_{\epsilon=\epsilon_u} = K\epsilon^n \Big|_{\epsilon=\epsilon_u}$$

$$\boxed{n = \epsilon_u}$$

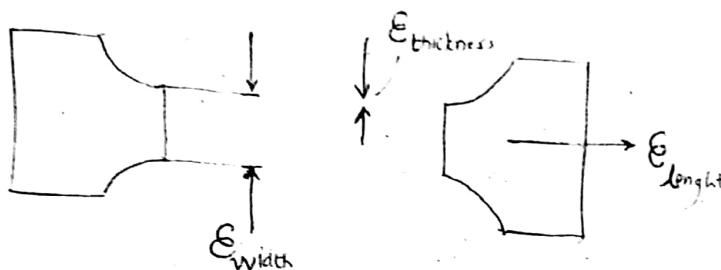
n value of materials = Uniform strength of materials

(Given material follows holloman hardening law)

$(n) \uparrow$ (strain(uniform)) \uparrow (ductility) \uparrow
 (Uniform elongation) \uparrow

* ANISOTROPY

Different properties in one direction



* Isotropic: $\epsilon_w = \epsilon_t$

* Volume constancy : $\epsilon_x + \epsilon_w + \epsilon_z = 0$

$$\Rightarrow \epsilon_x = -2\epsilon_w = -2\epsilon_t$$

$$\epsilon_w = \ln \frac{w}{w_0}$$

$$\epsilon_t = \ln \frac{t}{t_0}$$

* Lankford Anisotropy parameter = $R = \frac{\epsilon_w}{\epsilon_t} = \frac{\epsilon_w}{-(\epsilon_w + \epsilon_x)}$

⊗

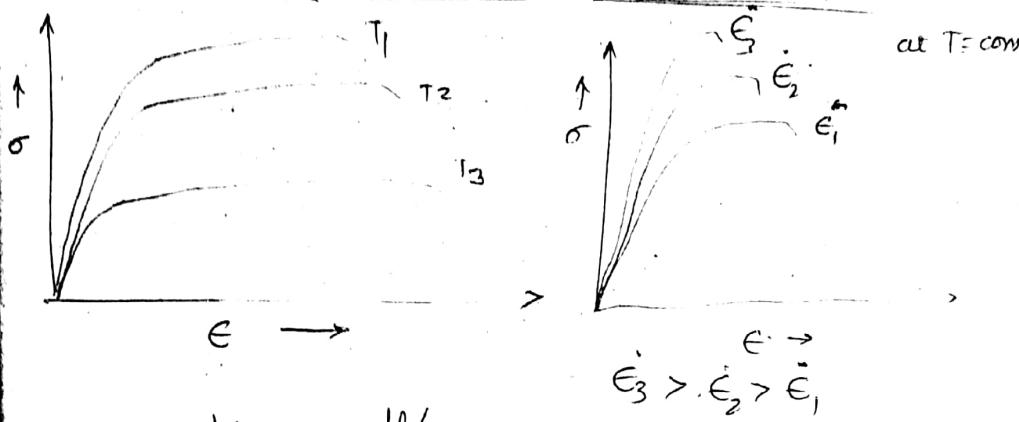
Transverse D

⇒ Lankford Anisotropy parameter R value
 determined along R_0 , R_{90} , R_{45} along rolling,
 transverse & 45° dir. respectively. (16)

$$\bar{R} = \text{Avg. Normal Anisotropy} = \frac{R_0 + 2R_{45} + R_{90}}{4}$$

$$\Delta R = \text{Planar Anisotropy} = \frac{R_0 - 2R_{45} + R_{90}}{2}$$

Effect of temperature & strain rate:



$$\dot{\epsilon} = \frac{d\epsilon}{dt} = \frac{dl/l}{dt}$$

ramp speed

$$= \frac{1}{l} \left(\frac{dl}{dt} \right) = \frac{u}{l}$$

UTM → constant ramp speed machine
 ↳ constant strain rate machine.

If a material strain changes w.r.t strain rate
 sensitive material.

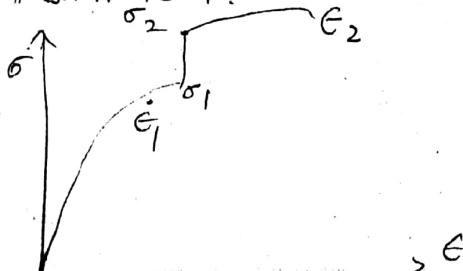
$$\boxed{\sigma = C \cdot \dot{\epsilon}^m}$$

where C = strain coeff

m = strain rate sensitivity index

$$m = \frac{\ln \left(\frac{\sigma_2}{\sigma_1} \right)}{\ln \left(\frac{\dot{\epsilon}_2}{\dot{\epsilon}_1} \right)}$$

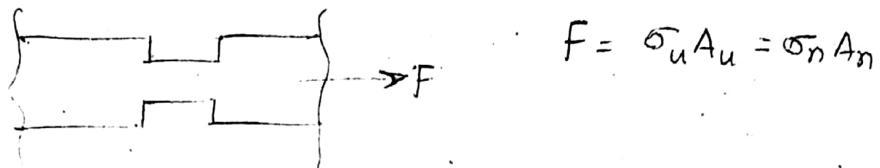
* JUMP TEST:



* Let us consider a specimen which has been already necked.

A_n = Area of c/s of neck

A_u = Area of c/s outside neck



$$\Rightarrow \frac{\sigma_u}{\sigma_n} = \frac{A_n}{A_u} \quad [A_s, \sigma = (\dot{\epsilon})^m]$$

$$\therefore \left(\frac{\dot{\epsilon}_u}{\dot{\epsilon}_n} \right) = \left(\frac{\sigma_u}{\sigma_n} \right)^{1/m} = \left(\frac{A_n}{A_u} \right)^{1/m}$$

Ques: $A_n = 90\% A_u$

Case 1: $m = 0.02$

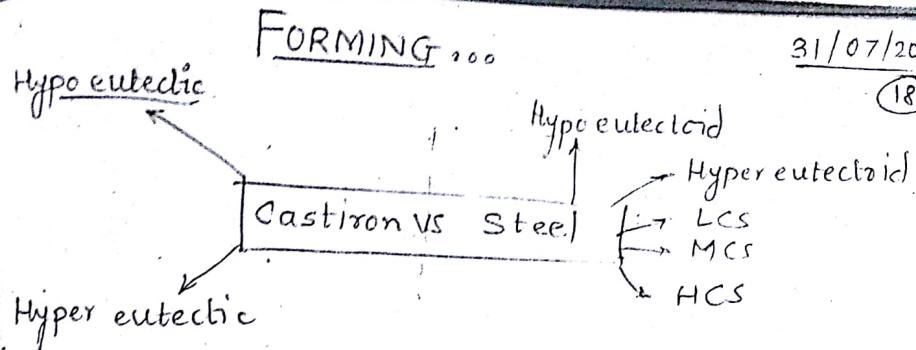
$$\frac{\dot{\epsilon}_u}{\dot{\epsilon}_n} = (0.9)^{1/0.02} = 5 \times 10^{-3}$$

Case 2 $\Rightarrow m = 0.5$

$$\frac{\dot{\epsilon}_u}{\dot{\epsilon}_n} = (0.9)^2 = 0.81$$

m = strain rate sensitivity index

$m \uparrow$ (post elongation(uniform)) \uparrow



Austenite
 ↓ ↓ ↓ ↓
 P S T B

- ④ $f(\text{temp}, \text{time})$
 ④ Complete transformation observed.

Austenite
↓
Martensite.

- $f(\text{temp.})$
complete transformation
not possible. some
amt. of A remains

INGOT → 1st product obtained.

Casting \Rightarrow INGOT \rightarrow Bloom / Billet
 primary forming - - - \downarrow

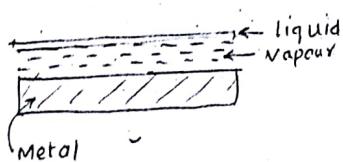
primary forming - - - ↓
Rolling process

secondary forming

secondary forming :
Sheet ← Slab { Temp ~

Coiled
[~600°C]

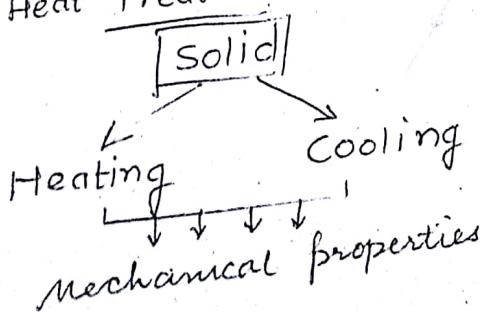
* Sudden cooling : (use of water jets)



Metal Run-out Table { cooling phenomenon }

Run-out : Heat Treatment Process :

Heat Treatment Process :

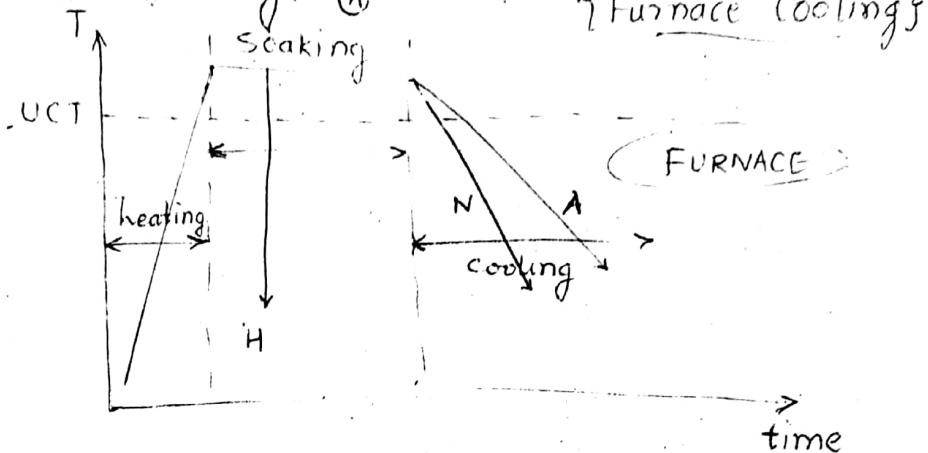


Heat Treatment.

- Reduce internal stress
- Refine grain structure
- Hardness, toughness, ductility.

Methods :

① [Annealing] \textcircled{A}



② [Normalizing] \textcircled{N}

{ Air Cooling }

• Cooling Rate (C.R)

Annealing < Normalising

• Product Hardness

Annealing < Normalising

③ [Hardening] \textcircled{H}

{ Water cooling }

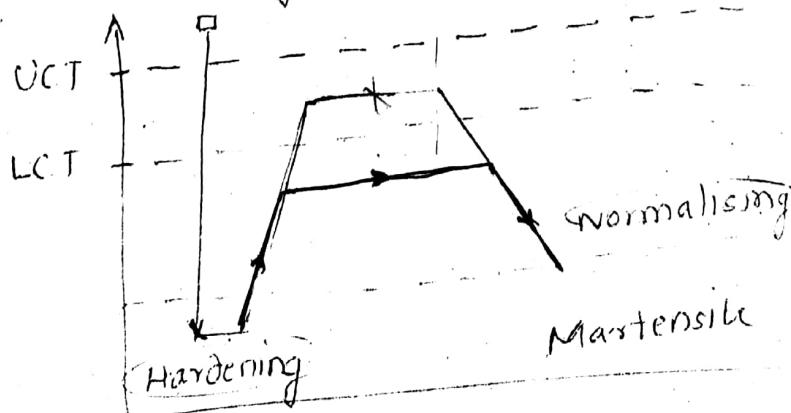
→ only incr. hardness, strength.

④ [Tempering]

→ Hardness + strength + Ductility + Tempering

↓
Hardness + Ductility

Hardening Normalising



(19) Surface Property : \rightarrow Case Hardening (20)

BOOKS :

- ① Physical Metallurgy - S.H. AVNER
- ② Manufacturing Science - GHOSH & MAHAK
- ③ Principles of Metal Working - G.W. ROWE

CASE HARDENING :-

Change of Surface Property

↓
Alloying (low carbon steel)

(1) Carburising @ 800°C - 900°C

(2) Nitriding @ $\sim 600^{\circ}\text{C}$

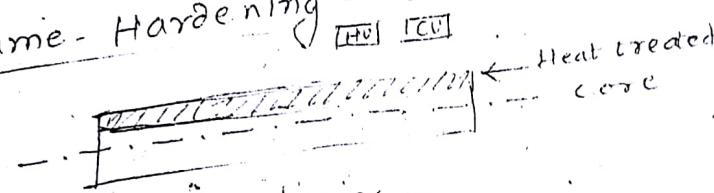
(presence of NH_3)

(3) Cyaniding $\sim 900^{\circ}\text{C}$

(dip in cyanide soln)

↓
Heat treatment
(medium carbon steel)

Flame Hardening - Use of oxyacetylene flame



Induction Hardening -

Laser Hardening

* Classification of Metal Working :-

HOT WORKING

Metal Working

COLD WORKING

$T_{\text{Working}} > T_{\text{Recrystallisation}} > T_w$

$T_{\text{Working}} \sim \frac{1}{3} \text{ or } \frac{1}{2} \text{ of m.p.t.}$

①