Integrating Materials and Manufacturing Innovation manuscript No. (will be inserted by the editor)

# Additive Manufacturing Melt Pool Prediction and Classification via Multifidelity Gaussian Process Surrogates

Robert Saunders $^{1,2}$  · Anna Rawlings $^1$  · Andrew Birnbaum $^1$  · Athanasios Iliopoulos $^1$  · John Michopoulos $^1$  · Dimitris Lagoudas $^2$  · Alaa Elwany $^2$ 

Received: date / Accepted: date

Abstract It is well known that the process parameters chosen in metal additive manufacturing (AM) are directly related to the melt pool dimensions, which can be related to microstructure characteristics, properties, and printability. Thus, the determination of melt pool dimensions and resulting printability, given a set of process parameters, is crucial to understanding the performance of AM parts. Unfortunately, experiments relating process parameters to the melt pool have a high execution and data collection cost. On the other hand, simulations do not suffer this cost penalty but can never be as good as the ground truth experiments. A number of capabilities exist to predict melt pool geometry from process parameters that range from highly accurate codes with long simulation times to less accurate analytical solutions that are nearly instantaneous. This work leverages multifidelity Gaussian process (GP) surrogates to examine how the fusion of information from different fidelities, including experiments, influences the resulting predictive model. Both a multifidelity GP regression and a novel multifidelity GP classification are examined. The multifidelity models are compared to standard GPs and a withheld set of test data. The results show good predictive accuracy for the melt pool width and depth, but suggest multifidelity models may not significantly improve classification of printability.

 $\textbf{Keywords} \hspace{0.2cm} \textbf{Metal Additive Manufacturing} \cdot \textbf{Melt Pool Prediction} \cdot \textbf{Printability} \cdot \textbf{Surrogate Modeling}$ 

Corresponding author: Robert Saunders

Tel.: +1-202-404-3489

E-mail: robert.saunders@nrl.navy.mil

 $<sup>^1\</sup>mathrm{U.S.}$ Naval Research Laboratory, Washington, DC, 20375

<sup>&</sup>lt;sup>2</sup>Department of Industrial & Systems Engineering, Texas A&M University, College Station, TX, 77840

## 1 Introduction

15

18

19

20

22

23

25

26

27

29

30

31

33

34

35

37

38

39

40

42

43

47

48

50

51

52

53

54

55

Additive manufacturing (AM) remains an area of critical technological and strategic importance to the United States government (USG) and its allies.[9] While AM has been actively researched as a science and engineering discipline for the past decade, its adoption as a mainstream manufacturing technology, particularly for metals, in practice has been slow, in part due to a lack of understanding of the performance of as-built AM parts and the challenges with qualification/certification that stem from this lack of understanding. [16] The Office of the President of the Unites States recently set out to address this lack of understanding in AM with AM Forward, a partnership between the USG and the manufacturing industry to conduct research to improve the performance of AM techniques. [2] While this effort primarily focuses on lowering the cost and increasing the quality and adoption of AM parts in 10 industry, there are additional USG-wide efforts to utilize AM technologies to create a more secure and 11 resilient supply chain. [1, 66, 67] In addition to these efforts, the US Department of Defense has identified AM as a crucial technology needing further development for US national security with the ability to 13 creater lighter, stronger, and more resillient components to better prepare and protect Warfighters. [65] 14

In order to improve the performance of metallic AM parts, the type of defects that arise during manufacturing and the causation mechanisms for the defects must be better understood. [6] The process to manufacture a metal AM part typically involves a metal powder being spread in a powder bed or deposited through a nozzle, and then an energy source (such as a laser or electron beam) is used to melt the powder in specific locations within a layer guided by a CAD file.[12] The molten material then cools and solidifies to create the desired geometry of a layer. This process is repeated until a full part is complete. The repeated thermal cycling of the powder and underlying solidified material in conjunction with the high thermal gradients and material phase transformation can lead to parts containing numerous defects that, ultimately, degrade the parts performance. [6] A key aspect to understanding the defects that are formed is understanding how the AM process parameters influence the resulting melt pool, or the liquid interface between the powder particles and the energy source. [36, 39] Too little energy being deposited will result in a melt pool not being formed, while a high energy will result in unsteady melt pool dynamics that can have unintended consequences.[16] In both cases, porosity defects will form. In the no melting case, a lack of fusion defect can form and lead to a lower effective material density part with irregular-shaped voids. In the high energy case, defects, such as keyholing and balling/beading, can lead to entrapped gas and again result in a part with a lower effective material density. In order to avoid these defects, the so-called *printability* needs to be assessed. Printability in this work is defined as the the process parameter map that can tell whether or not a porosity defect (i.e., lack of fusion, keyholing, balling/beading) will occur for a given set of process parameters. [3, 13, 23, 27, 52, 78] One popular way to asses and create this printability map is by *in-situ* process monitoring.[58] By monitoring the process and collecting the data, machine learning (ML) models can be trained to predict thermal history [41, 50, 68, 79], melt pool size and shape [52], anomalies [53], and printability [3, 78]. However, this work involves the continued collection of experimental data, which can be costly and time consuming, in order to train the ML models.

A less-costly alternative is to utilize one or more of the plethora of available modeling techniques to simulate every level of the AM process. [10, 35] Many of these models focus solely on predicting the thermal history and corresponding melt pool geometry as functions of input process parameters. [75] These process models range from analytical [8] and semi-analytical methods [55, 56, 70, 72, 76] to finite element/difference models [7, 15, 21, 22, 23, 32, 33, 49] to multi-physics models at the scale of the metal powder [11, 25, 26, 43, 44, 57, 73]. They generally fall into a spectrum from fast and approximate (e.g., analytical solutions) to slow and accurate (e.g., multiphyscics finite element or powder-scale models). In all cases, there will be some level of inaccuracy and uncertainty introduced by the modeling assumptions with higher fidelity models generally having fewer simplifying assumptions. [38] With lower fidelity models, uncertainty quantification (UQ) techniques can be used to interrogate the model relatively easily due to the low data acquisition cost. With higher fidelity models, this is not practical and many researchers implement machine learning (ML) or surrogate modeling techniques, such as Gaussian process (GP) models, to conduct UQ tasks, such as model calibration, uncertainty propagation, or sensitivity analysis. [19, 23, 34, 40, 59, 60, 61, 62, 69, 77] GPs, in general, are a popular tool to emulate AM process models as they limit the number of simulations required to effectively conduct UQ and can provide reasonably accurate approximations to the model they are emulating at a nearly negligible computational cost once trained.

A popular variation of the standard GP is the multi-fidelity GP developed by Kennedy and O'Hagan [24], commonly referred to as the Kennedy-O'Hagan or KOH model. In this framework, multiple sources of

information at varying degrees of fidelity can be incorporated into a single model. This limits the number of expensive simulations and/or experiments needed to build a model by leveraging information from cheap low fidelity solutions to the same problem. One example of this is using a few experiments to adjust a computer simulation.[17, 18] Recently, Mahmoudi *et al.* [34] applied this same methodology to a finite element (FE) AM process model to perform calibration of the model parameters and were able to achieve a high predictive accuracy.

In this work, a new multi-fidelity GP (MFGP) approach based on experiments and multiple fidelities of simulations is presented. Two MFGPs are shown, one that performs regression to determine melt pool dimensions and another that performs classification to determine printability. The MFGPs are trained using a combination of experiments and simulations for a model material of 316L stainless steel, and the data generation is scaled based on the fidelity (*i.e.*, a few experiments, but many analytical solution data points). In doing, so the MFGP is made to be as efficient as possible when compared to a standard single fidelity GP. The remainder of this work will be structured as follows: section 2 overviews the AM process models utilized to generate data; section 3 provides a brief overview of the MFGP regression along with a novel MFGP classification approach; section 4 explains the data generation process from the models as well as how the regression and classification models are trained; section 5 presents the validation of the MFGP against a test set of experiments, demonstrates how different combinations of fidelities can yield the best overall model, and analyzes how each process model contributes to the overall MFGP; and finally section 6 summarizes the work and provides prospects for future directions.

#### 77 2 AM Process Modeling

61

62

63

64

65

66

68

69

70

71

72

73

74

75

100

101

102

103

104

105

106

This work utilizes 4 different fidelity information sources to generate the data needed to train the MFGPs. 78 These are, namely, the analytical solution of Eager and Tsai [8], the Naval Reseach Laboratory (NRL) enriched analytical solution method [55, 56], a conduction-based FE model implemented in COMSOL 80 Multiphysics<sup>®</sup> [4], and laser powder bed fusion (L-PBF) experiments. The Eager-Tsai and NRL enriched 81 analytical solution method are both briefly detailed below and the interested reader is referred to the 82 referenced works for full details. The COMSOL FE model and experiments are detailed in full as they 83 are both previously unpublished. The computational models are setup to simulate the L-PBF process 84 with 316L stainless steel as used in the experiments. Additionally, each model takes as input the 316L 85 material properties and 3 process parameters, namely the laser power, velocity, and spot size<sup>1</sup>. The boundary value problem for the simulations is such that each considers a semi-infinite domain where the laser source is far from any boundaries. The top surface is a free surface and mirror symmetry is 88 implemented where applicable for half of the laser source and melt pool width. These conditions mimic those of the experiments where each single track is far from the base plate edges and, while multiple tracks are on a single plate, each is far from the others and a small dwell time is used to ensure temperature effects from previous tracks are not influencing the current track. Note that none of the simulations are 92 specifically calibrated to the experiments and only use material parameters obtained from literature for a 93 general 316L material. Temperature fields from each simulation are used to extract melt pool dimensions once a steady state thermal solution is achieved. Due to the high data acquisition and analysis cost of the higher fidelity information and limitations of the lowest fidelity model, only single-track simulations are considered to maintain consistency between the fidelities.

# 98 2.1 Eager-Tsai Analytical Solution

The Eager-Tsai (ET) analytical solution is the lowest fidelity source used in this work and calculates the temperature field due to a traveling Gaussian distributed heat source on a semi-infinite plate and uses the calculated temperature field to compute melt pool dimensions. The solution is based upon fundamentals of heat transfer, albeit with a number of simplifying assumptions, such as non-temperature dependent material properties and only heat conduction physics with no radiative or convection losses. The ET solution was originally developed to describe the impact of welding process parameters on the geometry and temperature distribution of weld melt pools.[54] However, it can be readily applied to develop a first order approximation for temperatures in the L-BPF process since L-PBF can be thought of as a repetitive micro-welding process.[20, 63, 78] While the ET solution is not highly accurate for predicting

 $<sup>^{1}</sup>$  More specifically, spot size is used to denote the distance equivalent to four standard deviations of a Gaussian beam profile, commonly referred to as D4 $\sigma$ 

all aspects of the L-PBF process, it is extremely fast and takes very little computational resources thus providing a good starting point for further analysis. A single simulation can run in seconds on a single if (10th generation) CPU in a laptop.

# 2.2 NRL Enriched Analytical Solution Method

The NRL Enriched analytical solution method (NEASM) was developed by modifying the ET solution 112 to incorporate several enrichments to better capture the features seen in the AM process. These are: 1) 113 using the linear heat equation to approximate the non-linear heat equation solution to account for the 114 temperature-dependent properties, via a fixed point iteration, 2) using the method of images to account 115 for finite domains, 3) inclusion of mass accretion by accounting for mass conservation, and 4) including 116 phase transformation. Enrichments 1 and 4 allow for temperature-dependent materials to be included 117 in the NEASM while enrichments 2 and 3 account for realistic part geometries and effects of multiple layers, respectively. In general, the NEASM is capable of predicting the thermal history of full AM part 119 builds including rastering within a layer and the effects of multiple layers. However, for this work, only 120 single-track data is needed thus it can be assumed that boundary effects are negligible (i.e., the laser is 121 sufficiently far from all boundaries and enrichment 2 is not necessary).[37] Additionally, under certain 122 circumstances, the NEASM can produce results with accuracy comparable to finite element modeling. [56] 123 Each simulation in this work using the NEASM takes under 1 minute on a single i7 (10th generation) 124 CPU in a laptop. 125

### 2.3 Finite Element Modeling

The finite element analysis (FEA) parametric simulation was implemented in COMSOL Multiphysics<sup>®</sup>
5.6[4] in conjunction with the LiveLink<sup>™</sup> capability enabling direct communication between COMSOL and
MATLAB<sup>®</sup> [64]. The MATLAB scripting capability offers the opportunity to automate the generation
of the datasets required to construct the surrogate models. The developed simulation capability is based
on an advective Eulerian approach that significantly reduces the computational cost as compared to a
traditional Lagrangian approach. The solved equation expresses the local balance of energy and reduces
to the steady-state form of the advective heat conduction equation:

$$\rho(T) C_p(T) \mathbf{u} \cdot \nabla T - \nabla \cdot \mathbf{q} = 0, \tag{1}$$

with T the temperature,  $\mathbf{q} = -k\left(T\right)\nabla T$ , the heat flux vector defined by the Fourier constitutive equation,  $\rho\left(T\right)$  the temperature dependent density,  $C_{p}\left(T\right)$  the temperature dependent heat capacity,  $k\left(T\right)$  the temperature dependent thermal conductivity,  $\mathbf{u} = \left\{-v\mathbf{i}, 0\mathbf{j}, 0\mathbf{k}\right\}^{T}$  the velocity vector, and v the velocity magnitude of the deposited beam power source in the x-direction (applied to the material domain while the heat source is maintained fixed in space).

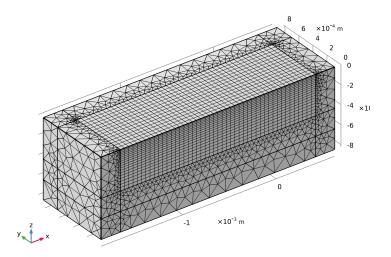


Fig. 1: One realization of the parameterized, discretized domain used in the COMSOL heat transfer simulations.

One realization of the discretized domain is shown in figure 1. The xz plane at y=0 is considered a symmetry plane by enforcing  $-\mathbf{n} \cdot \mathbf{q} = 0$ , with  $\mathbf{n}$  the normal vector of the surface. The domain is discretized with tetrahedral elements, utilizing the feature of COMSOL Multiphysics<sup>®</sup> for enabling semi-infinite domains with the intent to simulate an arbitrarily large geometries. A convective heat flux boundary condition was applied using  $-\mathbf{n} \cdot \mathbf{q} = h(T_{\infty} - T)$  to simulate the presence of material at the outside of these domains.  $T_{\infty}$  was taken as the room temperature. The value of h was set by simulating a complete domain with adequate material, such that  $T_{\infty}$  and T were equal, then comparing that simulation to a reduced domain that implemented the convective boundary conditions. After iterating, a value of h was determined that simulates the convection corresponding to the case with 316L material being the medium on the other side of the wall. The approximate value identified was  $h = 500 \, W/(m^2 K)$ . It should be noted that the results of the simulation are very insensitive to the actual value of h since the infinite element domain already addresses simulating boundary conditions at very large distances. The deposited beam power was applied at the top boundary of the hexahedral elements in the form of a heat flux boundary condition given by:

$$-\mathbf{n} \cdot \mathbf{q} = \alpha P_0 f(\mathbf{O}, \mathbf{e}) \frac{|\mathbf{e} \cdot \mathbf{n}|}{\|\mathbf{e}\|}, \tag{2}$$

with a = 0.45 the laser coupling coefficient,  $P_0$  the laser power and  $\mathbf{e} = \{0\mathbf{i}, 0\mathbf{j}, -1\mathbf{k}\}^T$  the beam orientation vector. The function f defines the deposited beam shape and was assumed to be of a Gaussian form according to:

$$f(\mathbf{O}, \mathbf{e}) = \frac{1}{2\pi\sigma^2} e^{-\frac{s^2}{2\sigma^2}},\tag{3}$$

use with

$$s = \frac{\|\mathbf{e} \times (\mathbf{x} - \mathbf{O})\|}{\|\mathbf{e}\|},\tag{4}$$

where x is the vector of coordinates of each boundary point and  $\mathbf{O}$  is the center of the beam application, considered here to be at point  $\{0,0,0\}^T$ . The Gaussian distribution variance was assumed to be equal to d/4 with d the apparent beam diameter, *i.e.*,  $\mathrm{D}4\sigma$  as defined previously.

One challenge with developing the automated dataset generation relates to the very wide extent of the parametric values, which entails different geometry sizes for different values of the parameters because of the different spread in temperature distribution. Since it is impossible to know a priori the extent of the melt pool, an iterative algorithm was developed to detect when the geometry needed to be adjusted to account for the size of the melt pool. The algorithm performs the simulation using nominal values, and if it identifies that the width or length of the melt pool is very close to the boundaries, it extends the domain and executes another simulation. The process is repeated until the computational domain is at least twice as big as the extent of the melt pool in all three directions.

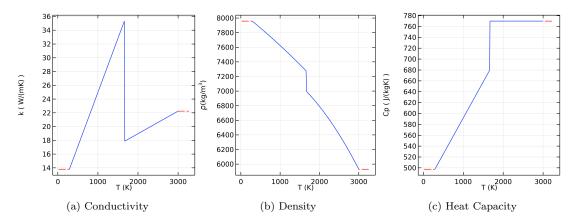


Fig. 2: The temperature dependent material properties assumed for the heat transfer problem.

The simulations were performed using the temperature dependent material properties shown in figure 2. The extent of the melt pool in the three directions representing width, length, and depth, were calculated by finding the extrema of the temperature iso-surfaces for  $T = T_{melt} = 1660K$ . Three simulations showing temperature distributions and domain size changes with parameters are shown in figure 3. The computational time for each of the simulations on a 4-core i7 (10th generation) laptop was at the level of 20-25s.

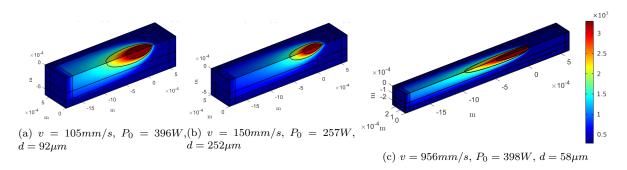
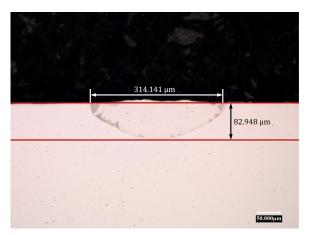


Fig. 3: Example temperature distributions for three different parameter combinations.

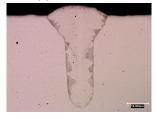
# 2.4 Experimental Data

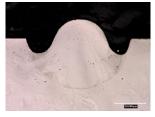
Experiment (EXP) data used for training and testing the MFGP scheme are single-track, so-called "bead-on-plate"/autogenous, experiments conducted on a GE/Concept Laser M2 selective laser melting (SLM) system on a 316L stainless steel plate.[48] The nominal processing conditions for the system are  $900\,mm/s$  laser velocity,  $370\,W$  laser power, and  $160\,\mu m$  laser  $(2\sigma)$  spot size. The minimum and maximum ranges of the system are  $10\text{-}7000\,mm/s$ ,  $75\text{-}400\,W$ , and  $50\text{-}350\,\mu m$ . These ranges are used to bound the data generation in the experiments as well as the other models with the exception that a maximum laser velocity of  $2000\,mm/s$  is set. This velocity was chosen as a practical maximum given the maximum possible power of the system *i.e.*, without an increase in power, such high velocities will seldom result in melted material.

Melt pool width and depth were extracted from the single-track experiments using optical microscopy. The samples were mounted and subjected to standard stainless steel grinding and polishing procedures (i.e., 320 grip paper, 9  $\mu m$  diamond polishing, 3  $\mu m$  diamond polishing, then 40 nm OP-S polishing). 10% oxalic acid electroetching at 5V for 15s was used to reveal the melt pool. Once mounted and processed, optical microscopy was performed. Melt pool width and depth were measured from the optical images. A representative output image from this process is shown in figure 4a In addition to melt pool width and depth, the melt pools were visually evaluated to determine if the run was in keyholing or conduction mode (figure 4b), whether beading (figure 4c), or if there was lack of fusion. Note that melt pool depth was measured from the surface in all cases for consistency.



(a) Measurements of melt pool width and depth.





(b) Melt pool image demonstrating keyholing

(c) Melt pool image demonstrating beading

Fig. 4: Representative single-track experiment SEM images.

# 3 Multi-fidelity Gaussian Processes

194

195

197

198

199

200

201

202

203

204

205

206

207

208

209

210

211

212

213

214

215

217

218

The KOH model [24] has been demonstrated to be very effective at multi-fidelity information fusion over the past two decades. However, it has two primary limitations. First, there is a linear correlation assumed between the higher fidelity data and the next lowest fidelity data. This linear correlation assumption is generally adequate but, in many computer models, this linear correlation only holds for specific ranges of the model. For instance, the ET model of this work always assumes conduction is the dominant heat transfer mechanism. When this is true, the ET model will likely be linearly correlated to higher fidelity models. However, at process parameters where conduction is not the primary mechanism, other models accounting for multiple heat transfer mechanisms could give a significantly different prediction that is not linearly correlated to the ET solution. The second limitation of the KOH model is that the computational complexity to train the model scales as  $\mathcal{O}((\sum_{t=1}^{s} n_t)^3)$  where s is the number of fidelities and  $n_t$  is the number of data points at each fidelity. This computational complexity can quickly become intractable considering that the lowest fidelity models can easily generate hundreds or thousands of points. However, both of the stated limitations have been overcome in recent years with Le Gratiet et al. [30, 31] addressing the latter and Perdikaris et al. [45] addressing the former. In the following derivations, standard GP regression and classification theory is briefly mentioned but for complete details, the interested reader is directed to the book of Rasmussen and Williams [47].

First, assume that there is a dataset  $\mathcal{D} = \{\boldsymbol{x}_i, y_i\} = (\boldsymbol{x}, y)$  for i = 1, ..., n with input vectors  $\boldsymbol{x}_i$  and responses  $y_i$  for n points. These data are assumed to be generated by some unknown latent function  $f(\cdot)$  which follows an n-dimensional multivariate Gaussian distribution such that,

$$p(f(\boldsymbol{x}_1), \dots, f(\boldsymbol{x}_n)) \sim \mathcal{N}_n(\boldsymbol{\mu}, \boldsymbol{k}),$$
 (5)

where  $\boldsymbol{\mu}$  is the mean vector defined by the mean function  $\mu(\boldsymbol{x}_i) = \boldsymbol{\mu}_i = \mathbb{E}\left[f(\boldsymbol{x}_i)\right]$  and  $\boldsymbol{k}$  is the covariance defined by the covariance function  $k(\boldsymbol{x}_i, \boldsymbol{x}_j) = \boldsymbol{k}_{ij} = cov\left[f(\boldsymbol{x}_i), f(\boldsymbol{x}_j)\right]$ . Now, the unknown latent function can be assigned a Gaussian process prior denoted as  $f(\cdot) \sim \mathcal{GP}(\mu(\cdot), k(\cdot, \cdot))$ . As is common, throughout this work it will be assumed that this and all other GP priors are zero mean i.e.,  $\mu(\cdot) = 0$ .

In the context of multi-fidelity modeling, there are now multiple datasets from each fidelity such that  $\mathcal{D}_t = (\boldsymbol{x}_t, y_t)$  for  $t = 1, \ldots, s$  with the s-level being the highest fidelity and the first level being the lowest.

In the KOH auto-regressive model, the s level are correlated as

220

221

222

224

227

229

230

237

240

241

246

247

249

252

254

255

$$f_t(\mathbf{x}) = \rho f_{t-1}(\mathbf{x}) + \delta_t(\mathbf{x}), \tag{6}$$

where  $\rho$  is a scaling factor that linearly correlates the t and t-1 fidelities and the bias of the lower fidelity is captured by  $\delta_t(\boldsymbol{x}) \sim \mathcal{GP}(0, k_t(\cdot, \cdot))$ . As mentioned above, the construction of the model in this way has a cost of  $\mathcal{O}((\sum_{t=1}^s n_t)^3)$  due to the required inversion of the covariance matrix to compute the posterior. Le Gratiet et al. derived a more numerically efficient scheme by replacing the GP prior  $f_{t-1}(\boldsymbol{x})$  with the corresponding posterior  $f_{t-1}^*(\boldsymbol{x})$  while maintaining  $\mathcal{D}_t \subseteq \mathcal{D}_{t-1}$ . Since nested data (i.e., data from fidelity t is a subset of the points in fidelity t-1) have been assumed, the posterior predictive density  $f_{t-1}^*(\boldsymbol{x})$  is deterministic at  $\boldsymbol{x}$ , which essentially decouples the MFGP problem into s standard GP problems. It was shown that this scheme results in the exact same posterior as the KOH model, while being more efficient (computational cost of  $\mathcal{O}(\sum_{t=1}^s \binom{n_t^3}{n_t^3})$ ) and yielding predictive models for each fidelity rather than only the highest fidelity as is the case in the KOH model.

To account for non-linear correlations, Perdikaris et al. modified equation 6 as

$$f_t(\mathbf{x}) = z_{t-1}(f_{t-1}(\mathbf{x})) + \delta_t(\mathbf{x}), \tag{7}$$

where  $z_{t-1}(\cdot)$  is an unknown function that maps the lower fidelity to the higher fidelity. This unknown function is assigned a GP prior such that  $z_{t-1}(\boldsymbol{x}) \sim \mathcal{GP}(0, k_t(\cdot, \cdot))$ . By assigning a GP prior to  $z_{t-1}(\cdot)$ , the posterior of  $f_t(\cdot)$  is no longer Gaussian and is considered a "deep GP". These deep GPs are generally very computationally complex, but can be made more tractable by following the scheme of Le Gratiet et al. where the prior of fidelity t-1 is replaced by the posterior as

$$f_t(\mathbf{x}) = z_{t-1}(f_{t-1}^*(\mathbf{x})) + \delta_t(\mathbf{x}) = g_t(\mathbf{x}, f_{t-1}^*(\mathbf{x})),$$
 (8)

where  $g_t(\mathbf{x}) \sim \mathcal{GP}(0, k_t((\mathbf{x}, f_{t-1}^*(\mathbf{x})), (\mathbf{x}', f_{t-1}^*(\mathbf{x}'))))$ . This formulation is made possible by the independence of  $z_{t-1}(f_{t-1}^*(\mathbf{x}))$  and  $\delta_t(\mathbf{x})$  as well as the fact that the sum of two GPs results in another GP. Perdikaris et al. note that this formulation retains the equivalent Markov property of the KOH model and the scheme of Le Gratiet et al.. Furthermore, with certain covariance kernel choices, the formulation of Le Gratiet et al. can be obtained. In this work, the covariance function of  $g_t$  takes the decomposed form of

$$k_{t_q} = k_{t_\rho}(\mathbf{x}, \mathbf{x}') \cdot k_{t_f}(f_{t-1}^*(\mathbf{x}), f_{t-1}^*(\mathbf{x}')) + k_{t_\delta}(\mathbf{x}, \mathbf{x}'),$$
 (9)

where each  $k_t$  is a valid covariance function. The covariance functions will take the common form of a stationary, squared exponential covariance as

$$k_t(\mathbf{x}, \mathbf{x}') = \eta_t \exp\left(-\frac{1}{2} \sum_{k=1}^p \theta_{k,t} (x_k - x_k')^2\right).$$
 (10)

When  $k_{t_{\delta}}(\boldsymbol{x}, \boldsymbol{x}')$  takes the form as specified, it will result in a nonlinear correlation, but in portions of this work, a linear correlation will be used for comparison and that will results in  $k_{t_{\delta}}(\boldsymbol{x}, \boldsymbol{x}')$  taking the form of a bias or constant kernel. The parameters  $\{\eta_t, \theta_{1,t}, \dots, \theta_{p,t}\}$  for  $t = 1, \dots, s$  make up the set of so-called hyper-parameters, which allow "tuning" of the correlation between data points and fidelities, and are learned from the data at each fidelity along with the posterior of the previous fidelity.

Note that up this point, no differentiation has been made between regression and classification. In both settings, the unknown latent function has the form as given and the posterior distribution can be found as

$$p(f_t^*(\mathbf{x}^*)) = \int p(f_t(\mathbf{x}^*, f_{t-1}^*(\mathbf{x}^*))|y_t, \mathbf{x}_t, \mathbf{x}^*) p(f_{t-1}^*(\mathbf{x}^*)) d\mathbf{x}^*.$$
(11)

In the standard GP regression problem and the non-linear auto-regressive GP (NARGP) regression with t=2, the likelihood is Gaussian and along with the assumed Gaussian priors, the posterior predictive distribution is Gaussian and the integral can be computed analytically. However, NARGP regression with t>2 and in all GP classification, the posterior is not Gaussian and alternative methods must be used to compute the integral to obtain the posterior predictive distribution.

#### 3.1 Regression

257

259

260

263

264

266

267

268

271

273

274

275

277

278

281

282

283

285

288

289

290

291

292

295

296

298

299

302

303

304

306

307

308

Even though the posterior for the NARGP with more than two levels is not Gaussian, the process to compute it is still quite simple. First, each level of the model is trained individually. At the first level, this corresponds to a standard GP regression with  $\mathcal{D}_1 = (\boldsymbol{x}_1, y_1)$ . Subsequent levels require modifying the training dataset such that  $\mathcal{D}_t = ((\boldsymbol{x}_t, f_{t-1}^*(\boldsymbol{x}_t)), y_t)$ . Acquiring the posterior  $f_{t-1}^*(\boldsymbol{x}_t)$  does not require approximating the integral of the previous fidelity, because the posterior of fidelity t-1 is deterministic at all points  $\boldsymbol{x}_t$  due to the assumed nested structure of the data. This training process is repeated for all fidelities until the s-levels are trained. Once trained, the posterior predictive density at a new point  $\boldsymbol{x}^*$  can be found at all fidelities t>2 by using Monte Carlo (MC) integration of equation 11. For fidelity 1 and 2, the posterior can be analytically obtained since it is Gaussian. Note that using MC integration requires the propagation of all MC samples from the previous fidelity to the current fidelity, while MC sampling again at the current fidelity for each of the MC samples generated at the previous fidelity. In doing so, the number of MC samples required to approximate the posterior scales exponentially and at the highest fidelity requires  $(n_{MC})^{s-2}$  total MC samples.

## 3.2 Classification

Unlike GP regression, the posterior for GP classification always requires approximation methods to evaluate the posterior. In GP classification, the latent function is considered a nuisance function where its value is never observed and, once a value is determined, it is transformed through a nonlinear warping to make predictions of observations. For the case of binary classification, which this work will focus on, this results in  $\pi(\mathbf{x}) \triangleq p(y=1|\mathbf{x}) = \Phi(f(\mathbf{x}))$ , where  $\pi(\cdot)$  is a deterministic function and, in this work,  $\Phi(\cdot)$ is the standard Normal cumulative distribution function i.e., a probit function is used for the warping. The probit function has the purpose of warping the latent function from  $[-\infty, \infty]$  to [0, 1], which is then used as a prediction of the class probability. Since this work focuses only on binary classification, the data likelihood will take the form of a Bernoulli distribution. [71] For the case of a single fidelity classifier, methods to approximate the posterior have been well studied and the approximation can be done via Laplace approximation or expectation propagation (EP) [47] or MC methods such as Markov chain Monte Carlo (MCMC). However, multi-fidelity GP classification has not been as well studied. Recently, Sahli Costabal et al. [51] developed and demonstrated a multi-fidelity GP classifier that was based on the KOH model and used MCMC to approximate the posterior. Likewise, Klyuchnikov and Burnaev [29] presented a Laplace approximation method for Gaussian process classification with two fidelities. However, this again was based on the KOH model using a linear correlation structure.

In this work, a novel approach to MFGP classification is taken. In general, all equations derived up to this point still hold. However, each dataset  $\mathcal{D}_t = (\boldsymbol{x}_t, y_t)$  now has labels  $y_t$  take the form of binary classes 0 or 1, rather than continuous variables. This allows the predictive posterior for the data to be written as

$$\pi_t^*(\mathbf{x}^*) \triangleq p(y^* = 1 | y_t, \mathbf{x}_t, \mathbf{x}^*) = \int \Phi(f_t^*(\mathbf{x}^*)) p(f_t^*(\mathbf{x}^*)) d\mathbf{x}^*.$$
 (12)

One can note that this posterior is the same as the posterior given in Rasmussen and Williams [47], except generalized to multiple fidelities and including a dependence (via equation 11) on the previous fidelity posterior prediction. By maintaining the assumptions given above, the problem of MFGP classification is reduced to fitting s single fidelity classifiers as developed by Le Gratiet et al. [30, 31] and allows the nonlinear correlation structure developed by Perdikaris et al. [45] to be maintained i.e., this approach is nearly identical to the NARGP regression above. The primary differences between the NARGP regression and classification are in training. First, to train each fidelity classifier, one must implement an approximation method as discussed above. In this work, that approximation is made using EP. Using an approximate inference method like EP is useful here as it allows the computational efficiency to be maintained. If implementing an MC type approach, sampling s fidelities can quickly become cumbersome and costly. From here, the approach to training is exactly as it is in the case of regression where each fidelity is trained recursively and the posterior of the latent function is used as an input to the next fidelity. It is important to emphasize here that, while the latent function in classification is unobserved, it does provide information about the underlying relationship between the data. If posterior predictive distribution for the data is used here as the input the the next fidelity, the warping function will tend to map points that could be very far apart so that they appear close together and related. Once the models are trained, MC integration is performed exactly as is done in the regression case except posteriors for

both the latent function and the data are needed. The latent function posterior samples are propagated to the next fidelity, while the posterior of the data is the prediction of the GPC for the current fidelity.

# 311 4 Data Generation & Model Training

312

313

314

315

316

317

318

319

320

321

322

323

324

326

327

328

329

331

332

333

334

335

336

337

339

340

341

In order to generate data to train the MFGP regression and classification, each of the four information sources detailed above must be sampled. The laser power (P), velocity (V), and spot size (S) are varied while the melt pool length (L), width (W), depth (D), conduction/keyhole mode class (K), beading class (B), and lack of fusion class (F) are output. The ranges for the input parameters are based on the experimental setup and are detailed above in section 2.4. Due to experimental limitation, melt pool length is not measured and is only available for simulations i.e., the 3 lowest fidelities. Conversely, the simulations do not contain sufficient physics to produce conduction/keyhole mode classifications. In lieu of direct class outputs for the simulations, empirical measures based on melt pool dimensions will be used to generate the needed output.[23] Since these empirical measures require melt pool dimensions, obtained through observation or regression, classifiers built using the empirical measures will be referred to as regression-based classifiers. In this work, keyholing will correspond to a melt pool width to depth ratio of W/D < 1.5. Additionally, lack of fusion and beading classification will be predicted as L/W > 2.3and D/t < 1, respectively, where t is the powder layer thickness, which will be 25  $\mu m$ . The set of these 3 classes will determine the so-called "printability" and give a predicted process parameter space where "good" (i.e., defect free) printing should occur. The framework for MFGP regression and classification is shown in figure 5.

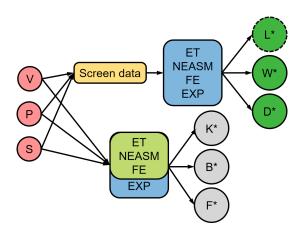


Fig. 5: MFGP regression and classification framework. Process parameter inputs are shown at the left, these are fed into the NARGP regression (possibly after some data screening, discussed later) and classification, which are used to create predictions of melt pool dimensions and printability. For regression, melt pool length is only trained for NARGP models without EXP. For classification, empirical classes are used for all simulation data.

In the case of a single fidelity data source, data is typically generated via a space-filling design of experiments (DoE) method such as Latin hypercube sampling (LHS). However, simply sampling each of the thermal models used here from the different fidelities would result in multiple space filling designs that would almost certainly not be nested, thus violating the assumptions of the theory above. With multi-fidelity modeling becoming more common in the past decade, numerous space filling nested DoE (nDoE) methods have been developed.[5, 14, 30, 31, 46, 74] In this work, the relatively simple scheme of Le Gratiet et al. [30, 31] is chosen. In this nDoE method, a space filling LHS design of  $n_s$  points is first created for the highest fidelity model. Subsequently, another LHS design is constructed for fidelity s-1 with  $n_{s-1}$  points. From the  $n_{s-1}$  points, the points that are closest in space to the higher fidelity  $n_s$  points are removed and replaced by the points  $n_s$ . In this work, a simple Euclidean distance measure is used but any desired distance measure could equivalently be implemented. The process as described is recursively applied until LHS designs are generated and modified for all s-fidelities. For this work, the ET model is sampled 500 times, the NEASM sampled 200 times, FE sampled 100 times, and experiments (henceforth referred to as EXP) sampled 50 times. For training, all data are used except 15 experiments,

which were randomly selected from the 50 available. The 15 experiments not used for training are used for validation/testing of the trained GPs.

The NARGP regression and classification as well as the nDoE are all implemented in Python 3.6.12. The NARGP regression and classification are based on Emukit (https://emukit.github.io/) [42], although the implementation has been modified to perform classification and cross-validation (although not used for this work) in the regression setting.

#### 5 Results & Discussion

In this work, the NARGP will be trained on all combinations of fidelities with 2 or more fidelities included. In addition, each of the combinations will be trained once with a nonlinear correlation between fidelities and a second time assuming a linear correlation between fidelities. This will result in 22 trained NARGP models (Table 1), which will be compared to standard GPs trained on a single fidelity. Each of the trained models will first be compared to the experiment test set and discussed, followed by a comparison of which fidelities included in the NARGP result in the best models for the least computational cost will be discussed.

Table 1: Multifidelity model combinations with corresponding labels of 1-22. Each design uses a binary labeling where a 1 indicates inclusion in the model and 0 exclusion. The nonlinear column specifies if a nonlinear correlation was used (1) or was not used (0)

	Nonlinear	$\mathbf{EXP}$	$\mathbf{FE}$	NEASM	$\mathbf{ET}$
1	1	1	1	1	1
2	0	1	1	1	1
3	1	0	1	1	1
4	0	0	1	1	1
5	1	1	0	1	1
6	0	1	0	1	1
7	1	0	0	1	1
8	0	0	0	1	1
9	1	1	1	0	1
10	0	1	1	0	1
11	1	0	1	0	1
12	0	0	1	0	1
13	1	1	0	0	1
14	0	1	0	0	1
15	1	1	1	1	0
16	0	1	1	1	0
17	1	0	1	1	0
18	0	0	1	1	0
19	1	1	0	1	0
20	0	1	0	1	0
21	1	1	1	0	0
22	0	1	1	0	0

# 5.1 Experiment Validation

# 5.1.1 Regression

The purpose of regression is to determine steady-state melt pool dimensions as a function of the AM process parameters. Since a classification is being trained to screen for defect inducing parameters and the purpose of the regression model is to predict good melt pool dimensions that can be propagated to subsequent models, the regression model can be trained in one of two ways. First, utilizing all data including data which produces melt pools that exhibit keyholing, beading, and/or lack of fusion. Or second, screen the training data beforehand and only training the regression model on the most relevant good print data. In the former case, more data will be available to train the model, but the data may be unnecessary since some regions will never be queried for predictions and won't necessarily reduce the uncertainty in relevant good regions. For the sake of comparison, models are trained using both data sets in this work. In both instances, a GP will also be trained on the highest fidelity model used in the NARGP. Melt pool length is not included here as melt pool length measurements were not able to

be collected in the experiments. Mean absolute percent error (MAPE) and coefficient of determination (R<sup>2</sup>) are used to evaluate the error of the trained model on the test data. The way in which MAPE is formulated can sometimes be problematic as it can result in erroneous, infinite, or undefined values.[28] As such, both R<sup>2</sup> and MAPE are used initially with R<sup>2</sup> being used as a verification of the MAPE results. The results of training can be seen in figures 6a and 6c and figures 6b and 6d for the models trained on all data and models trained on only good print data, respectively. Note, that in all cases, only test data that does not produce a defect will be used to evaluate the model. Since the regression model will only be used to predict good prints, these filtered test data will be the most relevant to evaluate true model accuracy. The filtered data set has 7 test points remaining of the original 15.

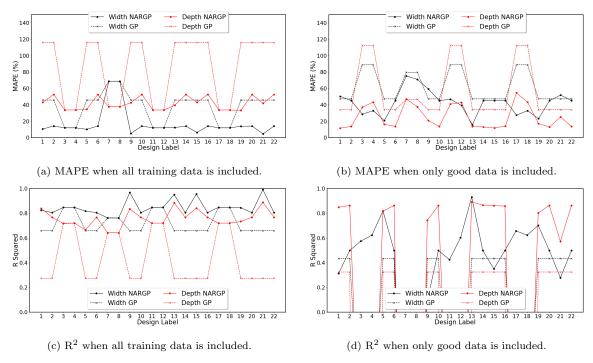


Fig. 6: MAPE and R<sup>2</sup> evaluated on test set melt pool width (W) and depth (D) data for the NARGPs of Table 1 and GPs.

From figures 6a and 6c, it can be seen that the GP can achieve a minimum error of 11.9% for the melt pool width and 33.7% for the depth when it is trained using the FE data. In general, the NARGP achieves a substantial improvement in accuracy, using both the MAPE and R<sup>2</sup> metrics, compared to the GP and in the worst cases, achieves a similar level of accuracy. In the best case scenario the NARGP can predict the melt pool width with 4.5% error and the depth with 33.7% error. For the melt pool width, the best model combinations were the EXP and FE with a nonlinear correlation structure, while for the depth, the best was any model trained using FE without experiments. The R<sup>2</sup> metric predicts a maximum melt pool width R<sup>2</sup> of 0.991 for the NARGP and 0.847 for the GP with a maximum melt pool depth R<sup>2</sup> of 0.889 for the NARGP and 0.72 for the GP. This again demonstrates that the NARGP produces a better model than the GP. The best melt pool width model using R<sup>2</sup> was again a combination of EXP and FE with a nonlinear correlation structure. For melt pool depth, the R<sup>2</sup> and MAPE metrics differ on which combination of models produce the best prediction indicating that there may be a few bad predictions of the melt pool depth that are skewing the MAPE metric. The highest R<sup>2</sup> value attained for melt pool depth was 0.889 for the combination of EXP and FE with a nonlinear correlation structure.

For the GP only fitted to good data (figures 6b and 6d), a similar level of accuracy as the GP trained on all data is achieved. The GP achieves a minimum error of 47.2% for the melt pool width and 33.9% for the depth when it is trained using the EXP data. As before and as expected, the NARGP improves on or maintains the accuracy and  $R^2$  values achieved by the GP alone with the best case predicting the melt pool width with 15.6% error ( $R^2=0.931$ ) using EXP and ET and a nonlinear correlation and the melt pool depth with 11.5% error ( $R^2=0.848$ ) using all fidelities and a nonlinear correlation. Similarly, a melt pool depth error of 11.6% ( $R^2=0.861$ ) can be achieved with the same fidelities as the previous

except excluding the ET data. Interestingly, the best  $R^2$  metric for melt pool depth of 0.891 was achieved using EXP and ET and a nonlinear correlation, which as before demonstrates that the best melt pool width and depth models determined by  $R^2$  were produced using the same combination of models.

Regardless of the removal of data before training, the NARGP produces better results than a GP trained on a single fidelity. Furthermore, the NARGP demonstrates its value over traditional multifidelity GP modeling approaches as the nonlinear correlation between fidelities produces the models with the highest accuracy. As one might expect, the NARGP including experiments produces the models with the lowest error and highest R<sup>2</sup> values. However, this is not the case for the GP where the model using FE data alone can produce a more accurate model (in percent error) than the one using experiments. This further highlights the need for multifidelity modeling to effectively increase the amount of information available for model training because, intuitively, experiments should always be used when available as they represent the ground truth. However, due to their high acquisition cost, only a limited number of experiments may be available and when taken alone may not produce the most accurate model as has been demonstrated.

Looking at the difference in the best cases when using all data versus only good data, it can be noted that the single best melt pool width prediction is obtained using all data, while the best melt pool depth prediction is obtained when using the screened good data considering MAPE alone as a metric. When  $\mathbb{R}^2$  is used an additional metric, the same conclusion can be drawn that the screened data generally produces a better model but the difference in the best result is not as noticeable e.g., the best  $\mathbb{R}^2$  for all data is 0.889 and for good data only, it is 0.891. This phenomenon can be attributed to how defects show up in the melt pool width and depth predictions. Melt pool width alone does not give a strong indication of a defect, unlike melt pool depth which can be a good indicator of keyholing and lack of fusion. Therefore, the melt pool width can be predicted more accurately when more data is available during training as is typically the case with GP-based models. For melt pool depth, the inclusion of data with defects and the fusion of information from models without the physics necessary to produce those defects tends to result in a model that is less accurate than a model trained on a consistent set of data i.e., one where simulations and experiments are both operating under the same physics. A similar trend on the difficulty of predicting melt pool depth can be seen in Tapia  $et\ al.$  [61].

In the best cases, an error rate of around 5-10% could be achieved in prediction (slightly under for 5% for melt pool width and slightly over 10% for melt pool depth). Likewise, a very strong correlation can be achieved for both the melt pool width ( $R^2 = 0.991$ ) and depth ( $R^2 = 0.891$ ). Depending on the application this could be an acceptable level of accuracy given the trade-off with computational time. Some recent works[19, 20, 34, 40] have demonstrated similar or slightly better accuracy on test data after training a GP model and using it to calibrate a physics-based model. Besides the obvious drawback of the increased computational time to query the model in comparison to the nearly negligible time to query a GP, this method has an additional drawback of a needing to use Bayesian method to perform the calibration. These Bayesian methods, while becoming more common, can be costly and require some subject matter expertise to obtain convergence. The method implemented here uses a more naive approach. While each model has the same general material of 316L stainless steel, none of the models have been specifically calibrated to produce the same results. Thus, for a new material system, one could simply obtain data and/or model parameters from literature or utilize existing models to generate a new predictive model without the need to perform a new calibration.

There are a number of possible improvements that could be made to the NARGP to perform regression, such as the inclusion of simulation data from a multiphysics model, but one of the most notable improvements would be to change the NARGP from having multiple independent outputs to being a multivariate or multi-output model[34]. The process to convert the NARGP to a multivariate model would be relatively straightforward, though still beyond the scope of this work. Since the NARGP essentially only requires the iterative training of multiple GPs, rather a single multifidelity GP, the process to convert to a multivariate model would only entail converting each GP into a multivariate GP and training the NARGP as usual. Converting to a multivariate model could accomplish two objectives. First, it could potentially improve the regression model by leveraging information from both width and depth simultaneously while training. Secondly, by having a multivariate model, Bayesian methods could be used to impute the missing melt pool length data when it is unavailable. The imputation of the missing data would be possible without a multivariate model, but with a multivariate model, the coupling of the melt pool length, width, and depth at lower fidelities could greatly improve the imputation. In addition to the improvements discussed on the GP modeling methodology, additional improvements may be possible by including additional process parameter inputs. However, given the accuracy achieved using only power, velocity, and spot size, it is unlikely that additional parameters would dramatically improve the results.

460

463

464

465

467

468

470

471

472

473

474

475

476

477

478

479

480

481

482

483

485

486

487

488

489

490

491

The purpose of classification is to determine the material printability map, in terms of porosity defect type, based on the AM process parameters. Unlike regression, classification will always be performed on all available data. For the sake of this work, classification will be conducted in a number of ways for comparison purposes. First, it is performed as described by Section 3.2 for each of the three defect classes, where each class is binary 1 for defect and 0 if that defect is not present. A second approach is to group all the defects into a binary problem of defect (1) or no defect (0) if one or more of the defects is present. Finally, an approach similar to previous approaches [23, 78] where a trained regression model and empirical measures, as detailed above, are used to predict if a defect will be present and what that defect will be. As the the beading metric requires melt pool length to approximate, any NARGP or GP that contains EXP will use the melt pool width and depth as normal but use melt pool length predictions based on the next lowest fidelity in the NARGP (e.g., an NARGP trained with EXP, FE, and NEASM will have melt pool length NARGP trained only on FE and NEASM and GP trained only on FE while EXP would be included in NARGP and GP for width and depth training). To evaluate each of the classification models, a balanced accuracy score (BAS) is used. The BAS is a standard measure of classification accuracy except each class is weighted by the number of samples in that class. The BAS is used in this work since the number of defects in the test set for each defect type is relatively low (3 of each type over the 15 total points in the test set). Note that the BAS is 1 for a perfect classifier and is bounded between 0 and 1.

The classification results when predicting each defect individually are shown in figure 7. In the case of beading (figure 7a), a maximum BAS of 0.875 is achieved and is obtained using the regression-based NARGP and GP. The NARGP is only able to gain this level of accuracy using only the NEASM and ET models with a linear correlation. The GP on the other hand obtains this accuracy for a model trained on the NEASM. A confusion matrix for this case is shown in table 2. Interestingly, the NARGP and GP classifiers can only obtain approximately 0.8 BAS. This is obtained for a number of combinations but most notably, none of the combinations involve using experiments. For keyholing (figure 7b), a maximum BAS of 0.83 is obtained with the NARGP classifier as well as regression-based classifier. For the classifier, this BAS is achieved using EXP, FE, and ET runs with a linear correlation and for the regression, it is achieved with EXP and NEASM runs using a non-linear correlation. The confusion matrices are identical for the two best classifiers and one is show in table 3. Interestingly, this maximum BAS regression case does not correspond to the regression results that best predict melt pool width and depth. In the case of lack of fusion (figure 7c), there a number of models that result in perfect predictions from all classifiers. The most notable result is with the NEASM and ET models combined or simply the NEASM, regardless of which method is used, can result in the highest BAS. Since the best cases result in perfect predictions, confusion matrices are not shown as they are simple matrices with zero as the off-diagonal components.

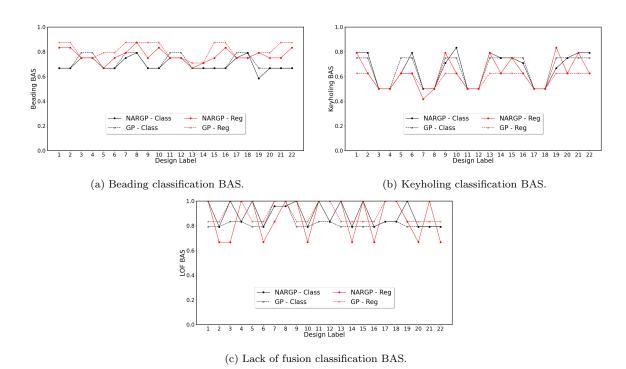


Fig. 7: Classification BAS for 3 defect types for the NARGP and GP classifiers and NARGP and GP regression-based classifiers.

Table 2: Confusion matrix for the best beading classification (GP regression-based classifier trained using NEASM).

	No Predicted Defect	Predicted Defect
No Actual Defect	9	3
Actual Defect	0	3

Table 3: Confusion matrix for the best keyholing classification (NARGP regression-based classifier trained using EXP and NEASM with a non-linear correlation).

	No Predicted Defect	Predicted Defect
No Actual Defect	12	0
Actual Defect	1	2

Again, note that the previous classifiers were tested on a set of 15 points where only 3 points for each of the defects were present in the 15. This results in highly skewed classes and makes determining the true model accuracy harder. By combining all defects into a single class, the skew becomes much less noticeable as there are 7 defects of the 15 points. Combining the classes also reduces training time as only a single model needs to be trained for the classifier. For the regression-based classifier, 3 models are still trained since all melt pool dimensions are still needed to make classifications before being combined into a single class. The results of the binary defect classification are shown in figure 8. The results show a highest possible BAS of around 0.8 for the regression-based classifiers. These occur for GPs trained on experiment data and NARGP and GPs trained on NEASM and ET and NEASM, as was seen before. As before, each of these best models results in an identical confusion matrix and one example is shown in table 4.

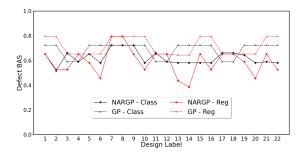


Fig. 8: Grouped binary classifier BAS for NARGP and GP classifiers and NARGP and GP regression-based classifiers.

Table 4: Confusion matrix for the best classification using the grouped binary classifier (GP regression-based classifier trained using EXP).

	No Predicted Defect	Predicted Defect
No Actual Defect	5	2
Actual Defect	1	7

504

506

507

508

510

511

512

513

514

515

517

518

519

520

521

522

524

525

526

527

528

529

530

532

533

534

A few interesting points can be noted about both the combined and individual defect classifiers. First, it appears that in many cases, the regression-based classifier (i.e., empirical classification measures from melt pool dimensions) shows similar or better performance than the regular classifiers. Furthermore, the NARGP classifier does not appear to outperform a standard GP classifier in most cases. This is in contrast to the regression where the NARGP could greatly outperform the standard GP. One possible explanation for the lack of performance of the NARGP classifier is that the BAS is only indicating the mean performance of the classifier and not the full probability space. In testing of the NARGP on a pedagogical example, it was noted that the NARGP classifier produced fewer correct classifications but the overall probability map was closer to the true boundary used to generate the data. It is possible that this is again occurring here, however, without a concrete means to assess this, the BAS and similar metrics are the only option to evaluate a model. Comparing the two approaches to classification, grouped versus individual, there seems to be little impact on the BAS results. In the individual cases, the NARGP and GP classifiers both perform around 0.8, which is close to the highest level of achievable accuracy in the grouped cases. However, this conclusion is somewhat superfluous as the regression-based classifier, using empirical measures derived from melt pool dimensions, appears to outperform the GP classifier, and the regression approach only makes predictions without considering individual versus grouped classes during training. An interesting point that was mentioned earlier is that a GP classifier trained on the NEASM as the highest or only fidelity tends to produce the best or nearly the best results in all cases except individual keyholing predictions, where the experiments produce the best model likely because the experiments are the only fidelity with the necessary physics to capture keyholing.

As with the regression there are some improvement that could be made to the classification. The most notable improvement would be to change to a multiclass classifier. This would nullify the need to decide between a binary defect/no-defect class or training individual classifiers. Much like the regression, changing to a multiclass classifier would be as simple as replacing the GP classifier implemented in this work with a new variant. The difficulty in changing to a multiclass classifier in the NARGP context is that the most common method to perform multiclass GP classification is via Bayesian methods, which become impractical when training multiple models sequentially. However, some recent works have begun to develop and implement non-Bayesian approaches to multiclass GP classification, which could be implemented in the NARGP framework. As with the NARGP regression problem, further improvements may be realized in the classification if additional input parameters are used that may help determine whether a defect will be induced or not. However, in the absence of additional data, especially from experiments, it is uncertain whether adding more inputs will improve the classification models or not.

#### 5.2 Process Model Importance

In the previous section, each of the 22 trained NARGPs was examined and compared individually to the test set of experiments. Now, the NARGP results will be used to determine what fidelities included in the results tend to result in a better performing model. This is done by grouping the MAPE and BAS for each NARGP that includes each fidelity. This is further split by separating the linear versus nonlinear correlation models. Box plots of the data generated by this process are shown in figure 9 for the melt pool width and depth MAPE (for models with all data used for training and only good data used) and in figure 10 for the grouped BAS from the NARGP classifier and NARGP regression-based classification. For brevity, only the grouped classification results are examined and R<sup>2</sup> is omitted since it generally agrees with the MAPE metric.

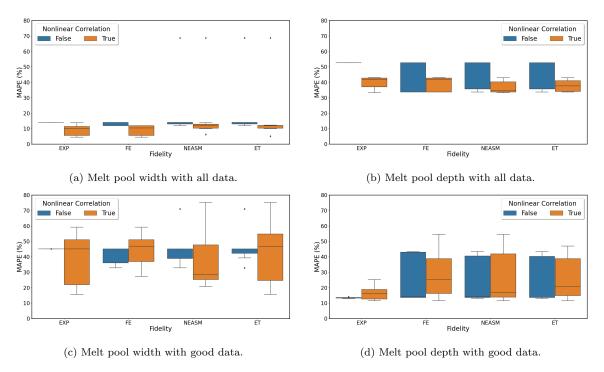


Fig. 9: Box plots showing MAPE ranges for NARGP models which included the given fidelity, both, with and without a nonlinear correlation.

In the case of the regression results for melt pool width, it can be noticed that a nonlinear correlation shows a wider spread of MAPE values, but tend to produce a better result in both cases of training with all data or only good data. In the case of the linear correlation, the melt pool width MAPE does not tend to change much as different fidelities are included in the NARGP. Whereas for the nonlinear correlation, the MAPE does tend to slightly decrease as higher fidelity data is added to the NARGP as one would expect. The results on melt pool width with all data versus good data also reinforce the conclusions above that a low MAPE can be achieved with a model trained on all data. With the exception of a couple outliers, the models trained on all data also show a lower variance than the models trained on good data only.

For melt pool depth, the conclusions from the previous section still hold, in that NARGP models trained on good data tend to produce lower error rates than a model trained on all data. Furthermore, the models trained using experiments result in the lowest MAPE values with the lowest variance. In the case of melt pool depth, the variance of linear correlation versus nonlinear correlation, as well as variance changes with fidelity, do not appear to change significantly until experiments are included. This, as stated, reduces the variance and tends to give the best results. Unlike melt pool width, the melt pool depth does not seem to improve significantly with higher fidelity data. The best model is any model multifidelity model that includes experiments. A similar conclusion can be made about melt pool width.

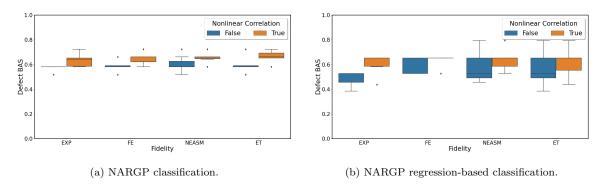


Fig. 10: Grouped defect BAS box plots for fidelities included in the NARGP model for models with and without nonlinear correlations.

Unlike regression, the classification shows consistent BAS values regardless of NARGP fidelities or correlation used. In fact, there is a slight upward trend in the BAS for the NARGP classifier when lower fidelities are included. A similar trend can be noted for the regression-based classifier, although the trend is not as prevalent. The trends shown in the box plots offer additional insight into the previous conclusion about the regression-based classification outperforming the standard classification. The NARGP classification, as detailed in this work, does tend to perform as well or better then the regression-based version. However, the highest BAS obtained is with the regression-based models. The classification method presented in this work also shows a lower variance, which could be a desirable feature that makes selection of specific fidelity data sets less important.

## 5.2.1 Model Selection

Having several fidelities of data available to make the best model is obviously ideal, but training all combinations of models and generating data from so many sources is generally not practical. Here some best practices are discussed that result in generating data from the fewest fidelities that result in the best trained models. This is done for regression only models, classification only models, and then models for classification and regression.

For regression, two models will likely need to be trained regardless; one for melt pool width on all available data and another for melt pool depth where the data has been screened a priori to select only data which does not produce defects. While training two models is not ideal, the benefits could be worthwhile since a melt pool width prediction of <5% and depth prediction of 11% is possible. Additionally, the NARGP is implemented with a maximum likelihood approach so training of a model is relatively fast. The drawback to achieve the level of accuracy shown is that at least 3 fidelities of data would be required, namely, EXP, FE, and NEASM. This could be a significant cost to acquire data from those sources and may not even be possible given that enriched analytical solutions are not widely available. Alternatively, a model using only EXP and ET data can be trained which results in a greatly reduced data acquisition cost with only a slight accuracy penalty, achieving 12.3% width and 13.6% error rates for the melt pool width and depth, respectively. While this level of error is somewhat high, it could serve as a good first order approximation and does not require specific calibration of the ET parameters using Bayesian methods.

In the case of classification, recommendations are much more difficult to make as clear trends are not seen as has been demonstrated in the previous sections. Using the NEASM model for any of the four classification methods shown (NARGP and GP classification and regression-based classification) appears to result in the best classifier for both individual and grouped classification except for keyholing predictions. Only models containing experiments are able to accurately predict keyholing to some extent. For the grouped classification, similar conclusions can be drawn. A GP regression-based classification with the NEASM or EXP provides the highest BAS around 0.8. Interestingly when combined in a multifidelity model, the result does not improve and tends to get slightly worse at 0.72 BAS.

Taking everything together, an NARGP model containing EXP, NEASM, and ET seems to be able to provide the best predictive model. The experiments improve keyholing and melt pool width and depth predictions, while the NEASM and ET have the purpose of predicting beading, lack of fusion, and/or grouped classification and add additional information to supplement the limited experiment data. While the resulting model would not be the best possible model, it would balance the data generation and

model creation processes. Additionally, it would only require training a single NARGP regression model.
While using the regression-based classification resulted in the best overall results, using the NARGP classification does have the potential to produce equivalent results to the regression-based version with less dependence on which fidelities are selected as was shown in the previous section.

#### 607 6 Conclusions

This work has presented an analysis and discussion on predicting and classifying AM melt pools using multifidelity GP surrogates. The work considered four common sources of information at different fidelities, from analytical to experimental, and combined that information in a NARGP regression and 610 novel classification models. The models were trained on a representative set of data and compared to a 611 test set of experiments. All combination of fidelities were trained and compared to a standard GP. The 612 NARGP regression demonstrated superior results over standard GP regression and require no model cal-613 ibration i.e., Bayesian methods. Furthermore, certain combinations of models, namely the experiments 614 and Eager-Tsai model, were able to produce an accurate model which balanced data generation cost 615 with attainable model accuracy. Depending on the fidelity of information available, a model was able to be produced, which matched or exceeded the performance of existing GP modeling and calibration 617 approaches. For classification, results using both grouped and ungrouped classes were examined along 618 with regression-based classification using empirical measures and how that compared to using binary 619 classification methods. The results showed that for ungrouped classification, experiments were necessary to accurately predict keyholing, as one would expect, but using only the NEASM or a multifidelity model 621 with the NEASM was sufficient to predict beading and lack of fusion. In the grouped classification set-622 ting, similar results were seen. In both cases, the regression-based classification was shown to produce 623 a singular best model for classifying printability, but typical classification methods were able to produce more consistent results that were less dependent on the fidelities included in the model. While the 625 NARGP classification approaches were able to produce some models that exceeded the performance of a 626 standard GP, the results were not consistent. It was postulated that while the NARGP did not improve 627 significantly on the classification accuracy for the available test set, it likely improved the predicted probability space. 629

Acknowledgements RS, AR, AB, AI, and JM acknowledge partial support for this work by the Office of Naval Research through the Naval Research Laboratory's (NRL) core funding. RS acknowledges partial support for this work by the NRL Edison Graduate Memorial Training Program.

#### 633 Conflict of interest

The authors declare that they have no conflict of interest.

#### References

640

641

- 1. Exec. order no. 14,017, 86 c.f.r. 11849. (2021).
- 2. Fact Biden administration celebrates launch of forward am and 637 congress pass bipartisan innovation act (2022).URL https: to 638 //www.whitehouse.gov/briefing-room/statements-releases/2022/05/06/
  - fact-sheet-biden-administration-celebrates-launch-of-am-forward-and-calls-on-congress-to-pass-b 3. Chen, Y., Wang, H., Wu, Y., Wang, H.: Predicting the printability in selective laser melting with a supervised machine learning method. Materials 13(22), 1-12 (2020). DOI 10.3390/ma13225063
- 4. COMSOL AB: Comsol multiphysics® v5.6. www.comsol.com. Stockholm, Sweden
- 5. Dash, S., Mandal, B.N., Parsad, R.: On the construction of nested orthogonal Latin hypercube designs. Metrika 83(3), 347-353 (2020). DOI 10.1007/s00184-019-00721-w. URL https://doi.org/10.1007/s00184-019-00721-w
- 6. DebRoy, T., Wei, H., Zuback, J., Mukherjee, T., Elmer, J., Milewski, J., Beese, A., Wilson-Heid,
  A., De, A., Zhang, W.: Additive manufacturing of metallic components Process, structure and
  properties. Progress in Materials Science 92, 112–224 (2018). DOI 10.1016/j.pmatsci.2017.10.001.
  URL https://linkinghub.elsevier.com/retrieve/pii/S0079642517301172

7. Denlinger, E.R., Gouge, M., Irwin, J., Michaleris, P.: Thermomechanical model development and in situ experimental validation of the Laser Powder-Bed Fusion process. Additive Manufacturing 16, 73-80 (2017). DOI 10.1016/j.addma.2017.05.001. URL http://dx.doi.org/10.1016/j.addma.2017.05.001

651

652

653

- 8. Eagar, T., TSAI, N.S.: Temperature Fields Produced by Traveling Distributed Heat Sources Use of a Gaussian heat distribution in dimensionless form indicates final weld pool shape can be predicted accurately for many welds and materials. Welding Journal (Miami, Fla) **62**(December) (1983)
- 9. Fast Track Action Subcommittee on Critical and Emerging Technologies: Critical and Emerging
  Technologies List Update. Tech. rep., National Science and Technology Counci (2022). URL http://www.whitehouse.gov/ostp.
- 10. Francois, M., Sun, A., King, W., Henson, N., Tourret, D., Bronkhorst, C., Carlson, N., Newman,
   C., Haut, T., Bakosi, J., Gibbs, J., Livescu, V., Vander Wiel, S., Clarke, A., Schraad, M., Blacker,
   T., Lim, H., Rodgers, T., Owen, S., Abdeljawad, F., Madison, J., Anderson, A., Fattebert, J.L.,
   Ferencz, R., Hodge, N., Khairallah, S., Walton, O.: Modeling of additive manufacturing processes for
   metals: Challenges and opportunities. Current Opinion in Solid State and Materials Science 21(4),
   198-206 (2017). DOI 10.1016/j.cossms.2016.12.001. URL http://dx.doi.org/10.1016/j.cossms.
   2016.12.001https://linkinghub.elsevier.com/retrieve/pii/S1359028616300833
- 11. Ganeriwala, R., Zohdi, T.I.: A coupled discrete element-finite difference model of selective laser sintering. Granular Matter **18**(2), 1–15 (2016). DOI 10.1007/s10035-016-0626-0
- 670 12. Gao, W., Zhang, Y., Ramanujan, D., Ramani, K., Chen, Y., Williams, C.B., Wang, C.C.,
  671 Shin, Y.C., Zhang, S., Zavattieri, P.D.: The status, challenges, and future of additive manu672 facturing in engineering. Computer-Aided Design 69, 65-89 (2015). DOI 10.1016/j.cad.2015.
  673 04.001. URL http://dx.doi.org/10.1016/j.cad.2015.04.001https://linkinghub.elsevier.
  674 com/retrieve/pii/S0010448515000469
- Gockel, J., Beuth, J.: Understanding Ti-6Al-4V microstructure control in additive manufacturing via
   process maps. 24th International SFF Symposium An Additive Manufacturing Conference, SFF
   2013 pp. 666–674 (2013)
- He, X., Qian, P.Z.: Nested orthogonal array-based Latin hypercube designs. Biometrika 98(3), 721–
   731 (2011). DOI 10.1093/biomet/asr028
- Heeling, T., Cloots, M., Wegener, K.: Melt pool simulation for the evaluation of process parameters in selective laser melting. Additive Manufacturing 14, 116–125 (2017). DOI 10.1016/j.addma.2017.
   02.003. URL http://dx.doi.org/10.1016/j.addma.2017.02.003
- 16. Herzog, D., Seyda, V., Wycisk, E., Emmelmann, C.: Additive manufacturing of met als. Acta Materialia 117, 371-392 (2016). DOI 10.1016/j.actamat.2016.07.019. URL
   http://dx.doi.org/10.1016/j.actamat.2016.07.019https://linkinghub.elsevier.com/
   retrieve/pii/S1359645416305158
- Higdon, D., Gattiker, J., Williams, B., Rightley, M.: Computer Model Calibration Using High Dimensional Output. Journal of the American Statistical Association 103(482), 570-583 (2008).
   DOI 10.1198/016214507000000888. URL https://www.tandfonline.com/doi/full/10.1198/
   016214507000000888
- 18. Higdon, D., Kennedy, M., Cavendish, J.C., Cafeo, J.A., Ryne, R.D.: Combining Field Data and Computer Simulations for Calibration and Prediction. SIAM Journal on Scientific Computing 26(2),
   448-466 (2004). DOI 10.1137/S1064827503426693. URL http://epubs.siam.org/doi/10.1137/S1064827503426693
- Honarmandi, P., Arróyave, R.: Uncertainty Quantification and Propagation in Computational Materials Science and Simulation-Assisted Materials Design. Integrating Materials and Manufacturing Innovation 9(1), 103-143 (2020). DOI 10.1007/s40192-020-00168-2. URL https://doi.org/10.1007/s40192-020-00168-2http://link.springer.com/10.1007/s40192-020-00168-2
- 20. Honarmandi, P., Seede, R., Xue, L., Shoukr, D., Morcos, P., Zhang, B., Zhang, C., Elwany, A.,
   Karaman, I., Arroyave, R.: A rigorous test and improvement of the Eagar-Tsai model for melt
   pool characteristics in laser powder bed fusion additive manufacturing. Additive Manufacturing
   47(August), 102300 (2021). DOI 10.1016/j.addma.2021.102300. URL https://doi.org/10.1016/j.addma.2021.102300
- Huang, Y., Yang, L.J., Du, X.Z., Yang, Y.P.: Finite element analysis of thermal behavior of metal
   powder during selective laser melting. International Journal of Thermal Sciences 104, 146–157
   (2016). DOI 10.1016/j.ijthermalsci.2016.01.007
- <sup>707</sup> 22. Hussein, A., Hao, L., Yan, C., Everson, R.: Finite element simulation of the tempera-<sup>708</sup> ture and stress fields in single layers built without-support in selective laser melting. Ma-

terials and Design 52(0), 638-647 (2013). DOI http://dx.doi.org/10.1016/j.matdes.2013.05. URL http://dx.doi.org/10.1016/j.matdes.2013.05.070http://www.sciencedirect. 710 com/science/article/pii/S0261306913005025

- 23. Johnson, L., Mahmoudi, M., Zhang, B., Seede, R., Huang, X., Maier, J.T., Maier, H.J., Karaman, 712 I., Elwany, A., Arróyave, R.: Assessing printability maps in additive manufacturing of metal alloys. 713 Acta Materialia 176, 199–210 (2019). DOI 10.1016/j.actamat.2019.07.005
- 24. Kennedy, M.: Predicting the output from a complex computer code when fast approximations are 715 available. Biometrika **87**(1), 1-13 (2000). DOI 10.1093/biomet/87.1.1. URL https://academic. 716 oup.com/biomet/article-lookup/doi/10.1093/biomet/87.1.1 717
- 25. Khairallah, S.A., Anderson, A.: Mesoscopic simulation model of selective laser melting of stainless steel powder. Journal of Materials Processing Technology 214(11), 2627–2636 (2014). DOI http: 719 //dx.doi.org/10.1016/j.jmatprotec.2014.06.001. URL http://www.sciencedirect.com/science/ 720 article/pii/S0924013614002234http://dx.doi.org/10.1016/j.jmatprotec.2014.06.001 721
- 26. Khairallah, S.A., Anderson, A.T., Rubenchik, A.M., King, W.E.: Laser powder-bed fusion ad-722 ditive manufacturing: Physics of complex melt flow and formation mechanisms of pores, spat-723 ter, and denudation zones. Acta Materialia 108, 36-45 (2016). DOI 10.1016/j.actamat.2016. 724 02.014. URL https://linkinghub.elsevier.com/retrieve/pii/S135964541630088Xhttp://dx. 725 doi.org/10.1016/j.actamat.2016.02.014 726
- 27. Khanzadeh, M., Chowdhury, S., Marufuzzaman, M., Tschopp, M.A., Bian, L.: Porosity prediction: 727 Supervised-learning of thermal history for direct laser deposition. Journal of Manufacturing Sys-728 tems 47(April), 69-82 (2018). DOI 10.1016/j.jmsy.2018.04.001. URL https://doi.org/10.1016/ j.jmsy.2018.04.001
- 28. Kim, S., Kim, H.: A new metric of absolute percentage error for intermittent demand forecasts. 731 International Journal of Forecasting 32(3), 669-679 (2016). DOI 10.1016/j.ijforecast.2015.12.003. 732 URL http://dx.doi.org/10.1016/j.ijforecast.2015.12.003 733
- 29. Klyuchnikov, N., Burnaev, E.: Gaussian process classification for variable fidelity data. Neurocom-734 puting **397**, 345-355 (2020). DOI 10.1016/j.neucom.2019.10.111. URL https://doi.org/10.1016/ 735 j.neucom.2019.10.111
- 30. Le Gratiet, L., Cannamela, C.: Cokriging-Based Sequential Design Strategies Using Fast Cross-Validation Techniques for Multi-Fidelity Computer Codes. Technometrics 57(3), 418–427 (2015). 738 DOI 10.1080/00401706.2014.928233. URL http://www.tandfonline.com/doi/full/10.1080/ 739 00401706.2014.928233 740
- 31. Le Gratiet, L., Garnier, J.: RECURSIVE CO-KRIGING MODEL FOR DESIGN OF COMPUTER EXPERIMENTS WITH MULTIPLE LEVELS OF FIDELITY. Interna-742 tional Journal for Uncertainty Quantification 4(5), 365–386 (2014). DOI 10.1615/Int.J. 743 UncertaintyQuantification.2014006914. URL http://arxiv.org/abs/1210.0686http://www.dl. begellhouse.com/journals/52034eb04b657aea,2f7b99cc281f2702,4c83626c5e64207a.html
- 32. Liu, Y., Zhang, J., Pang, Z.: Numerical and experimental investigation into the subsequent thermal 746 cycling during selective laser melting of multi-layer 316L stainless steel. Optics and Laser Technol-747 ogy 98, 23-32 (2018). DOI 10.1016/j.optlastec.2017.07.034. URL http://dx.doi.org/10.1016/j. optlastec.2017.07.034
- 33. Loh, L.E.E., Chua, C.K.K., Yeong, W.Y.Y., Song, J., Mapar, M., Sing, S.L.L., Liu, Z.H.H., Zhang, 750 D.Q.Q.: Numerical investigation and an effective modelling on the Selective Laser Melting (SLM) 751 process with aluminium alloy 6061. International Journal of Heat and Mass Transfer 80, 288-300 (2015). DOI 10.1016/j.ijheatmasstransfer.2014.09.014. URL http://dx.doi.org/10.1016/j. ijheatmasstransfer.2014.09.014 754
- 34. Mahmoudi, M., Tapia, G., Karayagiz, K., Franco, B., Ma, J., Arroyave, R., Karaman, I., Elwany, A.: 755 Multivariate Calibration and Experimental Validation of a 3D Finite Element Thermal Model for 756 Laser Powder Bed Fusion Metal Additive Manufacturing. Integrating Materials and Manufacturing 757 Innovation 7(3), 116-135 (2018). DOI 10.1007/s40192-018-0113-z. URL http://link.springer. 758 com/10.1007/s40192-018-0113-z
- 35. Markl, M., Körner, C.: Multiscale Modeling of Powder Bed-Based Additive Manufacturing. Annual Review of Materials Research 46(1), 93–123 (2016). DOI 10.1146/annurev-matsci-070115-032158. 761 URL http://www.annualreviews.org/doi/10.1146/annurev-matsci-070115-032158 762
- 36. Matthews, M., Roehling, T., Khairallah, S., Tumkur, T., Guss, G., Shi, R., Roehling, J., Smith, 763 W., Vrancken, B., Ganeriwala, R., McKeown, J.: Controlling melt pool shape, microstructure and residual stress in additively manufactured metals using modified laser beam profiles. Procedia CIRP 765 94, 200-204 (2020). DOI 10.1016/j.procir.2020.09.038. URL https://doi.org/10.1016/j.procir. 766

2020.09.038

- 37. Michopoulos, J.G., Steuben, J.C., Birnbaum, A.J., Iliopoulos, A.P., Aroh, J., Rollett, A.D., Gould,
   B.: Morphological Analysis of 316L Laser Powder Bed Fusion Melt-Pool via the Enriched Analytical Solution Method. In: Volume 9: 40th Computers and Information in Engineering Conference (CIE), pp. 1-14. American Society of Mechanical Engineers (2020). DOI 10.1115/DETC2020-22455. URL https://asmedigitalcollection.asme.org/IDETC-CIE/proceedings/IDETC-CIE2020/83983/Virtual,Online/1090045
- 774 38. Moges, T., Ameta, G., Witherell, P.: A Review of Model Inaccuracy and Parameter
  775 Uncertainty in Laser Powder Bed Fusion Models and Simulations. Journal of Man776 ufacturing Science and Engineering 141(4), 1 (2019). DOI 10.1115/1.4042789. URL
  777 http://manufacturingscience.asmedigitalcollection.asme.org/article.aspx?doi=10.
  778 1115/1.4042789https://asmedigitalcollection.asme.org/manufacturingscience/article/
  779 doi/10.1115/1.4042789/475003/A-Review-of-Model-Inaccuracy-and-Parameter
- 39. Mondal, S., Gwynn, D., Ray, A., Basak, A.: Investigation of melt pool geometry control in additive
   manufacturing using hybrid modeling. Metals 10(5), 1–23 (2020). DOI 10.3390/met10050683
- 40. Nath, P., Hu, Z., Mahadevan, S.: Uncertainty quantification of grain morphology in laser direct
   metal deposition. Modelling and Simulation in Materials Science and Engineering 27(4) (2019).
   DOI 10.1088/1361-651X/ab1676
- Ness, K.L., Paul, A., Sun, L., Zhang, Z.: Towards a generic physics-based machine learning model
   for geometry invariant thermal history prediction in additive manufacturing. Journal of Materials
   Processing Technology 302(August 2021), 117472 (2022). DOI 10.1016/j.jmatprotec.2021.117472.
   URL https://doi.org/10.1016/j.jmatprotec.2021.117472
- Paleyes, A., Pullin, M., Mahsereci, M., Lawrence, N., González, J.: Emulation of physical processes
   with Emukit. In: Second Workshop on Machine Learning and the Physical Sciences, NeurIPS (2019)
- 43. Panwisawas, C., Qiu, C., Anderson, M.J., Sovani, Y., Turner, R.P., Attallah, M.M., Brooks,
   J.W., Basoalto, H.C.: Mesoscale modelling of selective laser melting: Thermal fluid dynamics
   and microstructural evolution. Computational Materials Science 126, 479-490 (2017). DOI
   10.1016/j.commatsci.2016.10.011. URL http://dx.doi.org/10.1016/j.commatsci.2016.10.011
- 44. Pei, W., Zhengying, W., Zhen, C., Junfeng, L., Shuzhe, Z., Jun, D.: Numerical simulation and
   parametric analysis of selective laser melting process of AlSi10Mg powder. Applied Physics A:
   Materials Science and Processing 123(8), 1–15 (2017). DOI 10.1007/s00339-017-1143-7
- 45. Perdikaris, P., Raissi, M., Damianou, A., Lawrence, N.D., Karniadakis, G.E.: Nonlinear information
   fusion algorithms for data-efficient multi-fidelity modelling. Proceedings of the Royal Society A:
   Mathematical, Physical and Engineering Sciences 473(2198) (2017). DOI 10.1098/rspa.2016.0751
- 46. Qian, P.Z.: Nested Latin hypercube designs. Biometrika  $\bf 96(4),~957–970~(2009).$  DOI 10.1093/ biomet/asp045
- 47. Rasmussen, C.E., Williams, C.K.I.: Gaussian Processes for Machine Learning. The MIT Press (2006)
- 48. Rawlings, A.L.K., Birnbaum, A.J., Michopoulos, J.G., Steuben, J.C., Iliopoulos, A.P., Ryou, H.:
  Simulation Informed Effects of Solidification Rate on 316L Single Tracks Produced by Selective Laser Melting. In: Volume 9: 40th Computers and Information in Engineering Conference (CIE), February 2021. American Society of Mechanical Engineers (2020). DOI 10.1115/DETC2020-22451. URL https://asmedigitalcollection.asme.org/IDETC-CIE/proceedings/IDETC-CIE2020/83983/Virtual,Online/1090040
- 49. Roberts, I.A., Wang, C.J., Esterlein, R., Stanford, M., Mynors, D.J.: A three-dimensional finite element analysis of the temperature field during laser melting of metal powders in additive layer manufacturing. International Journal of Machine Tools and Manufacture 49(12-13), 916-923 (2009). DOI http://dx.doi.org/10.1016/j.ijmachtools.2009.07. 004. URL http://www.sciencedirect.com/science/article/pii/S0890695509001400http://dx.doi.org/10.1016/j.ijmachtools.2009.07.004
- 50. Roy, M., Wodo, O.: Data-driven modeling of thermal history in additive manufacturing. Additive Manufacturing 32(January), 101017 (2020). DOI 10.1016/j.addma.2019.101017. URL https://doi.org/10.1016/j.addma.2019.101017
- Sahli Costabal, F., Perdikaris, P., Kuhl, E., Hurtado, D.E.: Multi-fidelity classification using Gaussian processes: Accelerating the prediction of large-scale computational models. Computer Methods in Applied Mechanics and Engineering 357, 112602 (2019). DOI 10.1016/j.cma.2019.112602. URL https://doi.org/10.1016/j.cma.2019.112602
- 52. Scime, L., Beuth, J.: Using machine learning to identify in-situ melt pool signatures indicative of flaw formation in a laser powder bed fusion additive manufacturing process. Additive Manufacturing

- **25**, 151-165 (2019). DOI 10.1016/j.addma.2018.11.010. URL https://linkinghub.elsevier.com/retrieve/pii/S2214860418306869
- 53. Scime, L., Siddel, D., Baird, S., Paquit, V.: Layer-wise anomaly detection and classification for powder bed additive manufacturing processes: A machine-agnostic algorithm for real-time pixel-wise semantic segmentation. Additive Manufacturing 36(June), 101453 (2020). DOI 10.1016/j.addma. 2020.101453. URL https://doi.org/10.1016/j.addma.2020.101453

- 54. Seede, R., Shoukr, D., Zhang, B., Whitt, A., Gibbons, S., Flater, P., Elwany, A., Arroyave, R.,
   Karaman, I.: An ultra-high strength martensitic steel fabricated using selective laser melting additive manufacturing: Densification, microstructure, and mechanical properties. Acta Materialia 186,
   199-214 (2020). DOI 10.1016/j.actamat.2019.12.037. URL https://linkinghub.elsevier.com/retrieve/pii/S1359645419308766
- Steuben, J.C., Birnbaum, A.J., Iliopoulos, A.P., Michopoulos, J.G.: Toward Feedback Control for
   Additive Manufacturing Processes Via Enriched Analytical Solutions. Journal of Computing and
   Information Science in Engineering 19(3) (2019). DOI 10.1115/1.4042105
- 56. Steuben, J.C., Birnbaum, A.J., Michopoulos, J.G., Iliopoulos, A.P.: Enriched analytical solutions for additive manufacturing modeling and simulation. Additive Manufacturing 25(October 2018), 437–447 (2019). DOI 10.1016/j.addma.2018.10.017. URL https://doi.org/10.1016/j.addma.2018.
   10.017
- 57. Steuben, J.C., Iliopoulos, A.P., Michopoulos, J.G.: Discrete element modeling of particle-based additive manufacturing processes. Computer Methods in Applied Mechanics and Engineering 305,
   537-561 (2016). DOI 10.1016/j.cma.2016.02.023. URL http://dx.doi.org/10.1016/j.cma.2016.
   02.023
- 58. Tapia, G., Elwany, A.: A Review on Process Monitoring and Control in Metal-847 Based Additive Manufacturing. Journal of Manufacturing Science and Engineering 848 DOI 10.1115/1.4028540. **136**(6), 060801 (2014). URL http://manufacturingscience. 849 asmedigitalcollection.asme.org/article.aspx?doi=10.1115/1.4028540https:// asmedigitalcollection.asme.org/manufacturingscience/article/doi/10.1115/1.4028540/ 851 377521/A-Review-on-Process-Monitoring-and-Control-in
- 59. Tapia, G., Elwany, A.H., Sang, H.: Prediction of porosity in metal-based additive manufacturing using spatial Gaussian process models. Additive Manufacturing 12, 282–290 (2016). DOI 10.1016/j.addma.2016.05.009. URL http://dx.doi.org/10.1016/j.addma.2016.05.009
- 60. Tapia, G., Johnson, L., Franco, B., Karayagiz, K., Ma, J., Arroyave, R., Karaman, I., Elwany, A.: Bayesian Calibration and Uncertainty Quantification for a Physics-Based Precipitation Model of Nickel-Titanium Shape-Memory Alloys. Journal of Manufacturing Science and Engineering 139(7), 071002 (2017). DOI 10.1115/1.4035898. URL http://manufacturingscience.asmedigitalcollection.asme.org/article.aspx?doi=10.1115/1.4035898
- Tapia, G., Khairallah, S., Matthews, M., King, W.E., Elwany, A.: Gaussian process-based surrogate modeling framework for process planning in laser powder-bed fusion additive manufacturing of 316L stainless steel. International Journal of Advanced Manufacturing Technology 94(9-12), 3591–3603 (2018). DOI 10.1007/s00170-017-1045-z
- 62. Tapia, G., King, W., Johnson, L., Arroyave, R., Karaman, I., Elwany, A.: Uncertainty Propagation Analysis of Computational Models in Laser Powder Bed Fusion Additive Manufacturing Using Polynomial Chaos Expansions. Journal of Manufacturing Science and Engineering 140(12) (2018). DOI 10.1115/1.4041179. URL https:
  //asmedigitalcollection.asme.org/manufacturingscience/article/doi/10.1115/1.
  4041179/367000/Uncertainty-Propagation-Analysis-of-Computational
- 63. Tapia, G., King, W.E., Arroyave, R., Johnson, L., Karaman, I., Elwany, A.: Validation of a Laser-Based Powder Bed Fusion thermal model via Uncertainty Propagation and generalized Polynomial Chaos Expansions. Journal of Manufacturing Science and Engineering 140(December) (2018). DOI 10.1115/1.4041179. URL http://manufacturingscience.asmedigitalcollection.asme. org/article.aspx?doi=10.1115/1.4041179
  - 64. The Mathworks Inc.: Matlab® v2021b. www.mathworks.com. Natick, Massachusetts, USA
- 65. Under Sectrary of Defense for Research and Engineering: USD(R&E) Technology Vision for an Era of Competition. Tech. rep., US Department of Defese (2022)
- 66. U.S. Department of Defense: Securing defense-critical supply chains. Tech. rep. (2022)
- 67. U.S. Department of Energy: America's strategy to secure the supply chain for a robust clean energy transition. Tech. rep. (2022)

- 68. Wang, C., Tan, X.P., Tor, S.B., Lim, C.S.: Machine learning in additive manufacturing: State-of-the art and perspectives. Additive Manufacturing 36(January), 101538 (2020). DOI 10.1016/j.addma.
   2020.101538. URL https://doi.org/10.1016/j.addma.2020.101538
- 69. Wang, Z., Liu, P., Ji, Y., Mahadevan, S., Horstemeyer, M.F., Hu, Z., Chen, L., Chen, L.Q.: Uncertainty Quantification in Metallic Additive Manufacturing Through Physics-Informed Data-Driven
   Modeling. JOM 71(8), 2625-2634 (2019). DOI 10.1007/s11837-019-03555-z. URL https://doi.org/10.1007/s11837-019-03555-zhttp://link.springer.com/10.1007/s11837-019-03555-z
- 70. Weisz-Patrault, D.: Fast simulation of temperature and phase transitions in directed energy deposition additive manufacturing. Additive Manufacturing **31**(July 2019), 100990 (2020). DOI 10.1016/j.addma.2019.100990. URL https://doi.org/10.1016/j.addma.2019.100990
- 71. Williams, C., Barber, D.: Bayesian classification with Gaussian processes. IEEE Transactions on Pattern Analysis and Machine Intelligence **20**(12), 1342–1351 (1998). DOI 10.1109/34.735807. URL http://ieeexplore.ieee.org/document/735807/
- 72. Wolfer, A.J., Aires, J., Wheeler, K., Delplanque, J.P., Rubenchik, A., Anderson, A., Khairallah, S.:
  Fast solution strategy for transient heat conduction for arbitrary scan paths in additive manufacturing. Additive Manufacturing 30(October), 100898 (2019). DOI 10.1016/j.addma.2019.100898. URL https://doi.org/10.1016/j.addma.2019.100898
- 73. Xia, M., Gu, D., Yu, G., Dai, D., Chen, H., Shi, Q.: Porosity evolution and its thermodynamic mechanism of randomly packed powder-bed during selective laser melting of Inconel 718 alloy. International Journal of Machine Tools and Manufacture 116(January), 96–106 (2017). DOI 10.1016/j.ijmachtools.2017.01.005. URL http://dx.doi.org/10.1016/j.ijmachtools.2017.01.005
- 74. Xu, J., Duan, X., Wang, Z., Yan, L.: A general construction for nested Latin hypercube designs.
   Statistics and Probability Letters 134(11771450), 134-140 (2018). DOI 10.1016/j.spl.2017.10.022.
   URL https://doi.org/10.1016/j.spl.2017.10.022
- 75. Yan, Z., Liu, W., Tang, Z., Liu, X., Zhang, N., Li, M., Zhang, H.: Review on thermal analysis in laser-based additive manufacturing. Optics and Laser Technology **106**, 427–441 (2018). DOI 10.1016/j.optlastec.2018.04.034. URL https://doi.org/10.1016/j.optlastec.2018.04.034
- 76. Yang, Y., van Keulen, F., Ayas, C.: A computationally efficient thermal model for selective laser melting. Additive Manufacturing 31(October 2019), 100955 (2020). DOI 10.1016/j.addma.2019. 100955. URL https://doi.org/10.1016/j.addma.2019.100955
- Ye, J., Mahmoudi, M., Karayagiz, K., Johnson, L., Seede, R., Karaman, I., Arroyave, R., Elwany,
   A.: Bayesian Calibration of Multiple Coupled Simulation Models for Metal Additive Manufacturing:
   A Bayesian Network Approach. ASCE-ASME J Risk and Uncert in Engrg Sys Part B Mech Engrg
   8(1), 1-12 (2022). DOI 10.1115/1.4052270. URL https://asmedigitalcollection.asme.org/risk/article/8/1/011111/1119199/Bayesian-Calibration-of-Multiple-Coupled
- 78. Zhang, B., Seede, R., Xue, L., Atli, K.C., Zhang, C., Whitt, A., Karaman, I., Arroyave, R., Elwany,
  A.: An efficient framework for printability assessment in Laser Powder Bed Fusion metal additive manufacturing. Additive Manufacturing 46(June), 102018 (2021). DOI 10.1016/j.addma.2021.
  102018. URL https://doi.org/10.1016/j.addma.2021.102018https://linkinghub.elsevier.com/retrieve/pii/S2214860421001834
- 79. Zhou, Z., Shen, H., Liu, B., Du, W., Jin, J.: Thermal field prediction for welding paths in multi-layer gas metal arc welding-based additive manufacturing: A machine learning approach. Journal of Manufacturing Processes 64(February), 960-971 (2021). DOI 10.1016/j.jmapro.2021.02.033. URL https://doi.org/10.1016/j.jmapro.2021.02.033