RESEARCH ARTICLE

Double-layered TiO₂ cavity/nanoparticle photoelectrodes for efficient dye-sensitized solar cells

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ABSTRACT: TiO₂ nanoparticles (NPs) in the size of ~25 nm, namely P25, are very common material as the electron collecting layer in dye-sensitized solar cells (DSSCs). However, the light-scattering improvement of TiO₂ NP photoelectrodes is still a challenge. Here, we built TiO2 cavities on the top of the TiO2 NP layer by using carbonaceous microspheres as the template, forming the TiO₂ cavity/nanoparticle (C/NP) photoelectrode for the application in DSSCs. The cavity amount in the TiO2 C/NP photoelectrode was controlled by adjusting the weight ratio of carbonaceous microspheres. SEM results confirm the successful formation of the double-layered TiO₂ C/NP electrode. J-V tests show that the optimized TiO₂ C/NP electrode prepared with 25 wt.% carbonaceous microspheres contributes to remarkable improvement of the short-circuit current density (J_{sc}) and the power conversion efficiency (PCE). The best photovoltaic performance solar cell with the PCE of 9.08% is achieved with the optimized TiO₂ C/NP photoelectrode, which is over 98% higher than that of the TiO₂ NP photoelectrode. Further investigations of UV-vis DRS, IPCE, OCVD, and EIS demonstrate that the competition between light scattering effect and charges recombination in this TiO₂ C/NP photoelectrode is responsible for the PCE enhancement.

KEYWORDS: titanium dioxide; dye sensitized solar cell; cavity; light scattering

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Received August 22, 2022; accepted January 8, 2023

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1 Introduction

As one of the most competitive types of the thirdgeneration solar cells, dye-sensitized solar cells (DSSCs) have become a hotspot in the field of solar cell research due to their several advantages including the easy fabrication process, the use of nontoxic materials, and the designability of flexible devices [1–2]. DSSCs are typically assembled with three parts, including organic dye-sensitized electron transport layer built by TiO₂ nanoparticles (NPs) in the size of ~25 nm (known as P25) film as the photoelectrode, I⁻/I₃⁻ solution as the redox electrolyte, and Pt-coated fluorine-doped tin oxide (FTO) glass as the counter electrode [3]. So far, investigators have done much research work to optimize the photovoltaic performance of DSSCs, such as synthesizing organic efficient dyes [4-5],designing microstructures of TiO₂ [6-7], and developing low-price counter electrode materials to replace the conventional expensive Pt [8-9]. As is well known, the TiO₂ NP photoelectrodes are most widely used in DSSCs because of their high specific area for the dye adsorption [10–11]. However, the TiO₂ NP photoelectrodes always experience bad light scattering because of small NPs and 5-15 nm pores in the TiO2 NP layer [12]. A strategy building the light-scattering center or layer could be a solution to overcome this drawback [13].

Generally, two approaches could be performed to generate light-scattering effects in the photoelectrode. One is the construction of large-size TiO₂ structures, such as TiO_2 beads [14–16], microspheres [17–19], hollow microspheres [20-22], nanotubes [23-24], and nanorods [25–26]. Another approach could be the building of TiO₂ cavities in the photoelectrode. Microsphere templates such as polystyrene spheres or carbon spheres are suitable candidates to create cavities in the photoelectrode [27]. During the sintering process, such templates embedded in the TiO₂ NP photoelectrode would be sacrificed, leaving cavities which function as light-scattering centers in the photoelectrode [12,28]. Nevertheless, cavities in the TiO₂ NP photoelectrode may destroy the mesoporous scaffold and prolong the pathway of electrons. Meanwhile, building cavities in the TiO₂ NP photoelectrode is still an attractive strategy because of its low synthesis cost and easy fabrication process.

In this study, we prepared carbonaceous microspheres in the size of ~500 nm and used them as the template to make the TiO₂ cavity layer on the top of the TiO₂ NP photoelectrode prepared by P25, forming the TiO₂ cavity/nanoparticle (C/NP) photoelectrode. The weight ratio of carbonaceous microspheres was set in the range of 0–50 wt.% to build double-layered TiO₂ NP and TiO₂ C/NP photoelectrodes with a controlled amount of cavities. Based on such photoelectrodes, DSSCs were assembled and their photovoltaic performance was measured. The best power conversion efficiency (PCE) of

9.08% was obtained with an optimized TiO₂ C/NP photoelectrode, which is over 98% higher than that of the TiO₂ NP photoelectrode. Further investigations demonstrated that the increase in the photovoltaic performance of this optimized TiO₂ C/NP-based DSSC can be attributed to the competition between light scattering effect and charges recombination in the TiO₂ C/NP photoelectrode.

2 Experimental

2.1 Materials

Analytical pure chemical reagent, including sucrose, ethylcellulose, terpineol, and ethanol, were ordered from Aladdin Co., Ltd. (Shanghai, China). These chemicals were used directly without further purification. Other materials such as commercial P25, conductive FTO glass, organic dye di-tetrabutylammonium *cis*-bis(isothiocyanato)-bis(2,2'-bipyridyl-4,4'-dicarboxylato)-ruthenium(II) (abbreviated as N719), Pt/FTO counter electrode, and I⁻/I₃⁻ electrolyte were offered by Opvtech Co., Ltd. (Dalian, China).

2.2 Building of TiO₂ NP layer on FTO

The slurry composed of $0.2 \, \mathrm{g}$ ethylcellulose, $3 \, \mathrm{mL}$ terpineol, $2 \, \mathrm{mL}$ ethanol, and $1.0 \, \mathrm{g}$ P25 was spin-coated on the conductive surface of the FTO glass which was cut in advance into the size of $2.0 \, \mathrm{cm} \times 1.5 \, \mathrm{cm}$. Then the product was transferred into a Muffle furnace and sintered at $450 \, ^{\circ}\mathrm{C}$ for $30 \, \mathrm{min}$ to burn out organic residues, and finally the TiO_2 NP electrode was produced.

2.3 Formation of TiO₂ cavities on the top of TiO₂ NP layer

Carbonaceous microspheres were firstly synthesized as the hard template according to Ref. [29] for making cavities in TiO_2 NPs. Briefly, the 1.5 $\text{mol} \cdot \text{L}^{-1}$ sucrose aqueous solution was added into 100 mL stainless autoclave with 80% volume and sealed. Subsequently, the stainless autoclave was put into an electric oven to complete the hydrothermal reaction. The hydrothermal reaction condition was set to 160 °C for 8 h.

Carbonaceous microspheres, P25, ethylcellulose (0.2 g), terpineol (3 mL), and ethanol (2 mL) were mixed together to get a homogenous viscous mixture. Then, the viscous

mixture was doctor-bladed on the top of the prepared TiO₂ NP electrode. After dried in air, the product was sintered again in the Muffle furnace at 450 °C for 1 h to burn out carbonaceous microspheres, leaving cavities inside the top layer of the TiO2 NP electrode. The resultant photoelectrode is hereafter abbreviated as TiO2 C/NP. The amount of cavities in TiO₂ C/NP is tuned by the weight ratio of carbonaceous microspheres in the mixture. We prepared TiO₂ C/NP with the weight ratio of carbonaceous microspheres in 12.5, 25, and 37.5 wt.%, which are abbreviated as TiO₂ C/NP (12.5 wt.%), TiO₂ and TiO_2 C/NP (37.5 wt.%), C/NP (25 wt.%),respectively. A double-layered TiO₂ NP photoelectrode was also prepared as a comparative study.

2.4 Dye sensitization and solar cell assembly

The 0.3 mmol·L⁻¹ N719 ethanol solution was used to complete the dye loading on the photoelectrode. Specifically, the series of TiO₂ C/NP electrodes and the TiO₂ NP electrode were dipped into the N719 solution for 72 h at room temperature. Then, these photoelectrodes were rinsed with ethanol and dried in air.

The DSSC was assembled in an open way like a sandwich by inserting the I^-/I_3^- electrolyte between the dye-sensitized photoelectrode and the Pt/FTO counter electrode [30].

2.5 Characterization

The morphology and the structure of carbonaceous microspheres and photoelectrodes (i.e., TiO₂ NP and TiO₂ C/NP) were characterized by field-emission scanning electron microscopy (FE-SEM, FEI-Quanta 250) and transmission electron microscopy (TEM, FEI Tecnai G2). A Quantachrome NOVA 4200e system was used to test nitrogen adsorption—desorption isotherms. Before the test, samples were degassed at 250 °C for 12 h under vacuum. Based on nitrogen adsorption—desorption isotherms, the specific surface areas of TiO₂ C/NP photoelectrodes were calculated according to the Brunar—Emmett—Teller (BET) method. The diffuse reflectance spectroscopy (DRS) results of those photoelectrodes were obtained by an ultraviolet-visible (UV-vis) spectrophotometer (Hitachi U4100).

J-V behaviors of the DSSCs were analyzed by a Keithley 2400 source meter under AM 1.5 simulated sunlight with a power density of $100 \text{ mW} \cdot \text{cm}^{-2}$. The

of the incident measurements photon-to-current conversion efficiency (IPCE) spectrum were conducted by the 150 W Xe lamp coupled with a computer-controlled monochromator. The open-circuit voltage decay (OCVD) measurement and the electrochemical impedance spectroscopy (EIS) were carried out on electrochemical workstation CHI 660E. The DSSCs for the OCVD tests were illuminated under AM 1.5 simulated sunlight, and the variation of voltage with time was recorded after turning off the simulated sunlight. The EIS was performed under dark at a bias of -0.72 V with the frequency range from 0.1 to 10⁵ Hz.

3 Results and discussion

The carbonaceous microspheres play the role of template to make cavities in the TiO₂ NP photoelectrode. Figure 1(a) shows the SEM image of carbonaceous microspheres prepared by the hydrothermal method using the sucrose aqueous solution as the raw material. Apparently, microspheres with the size around 500 nm are successfully obtained. The TEM image of carbonaceous microspheres is also presented in Fig. 1(b), indicating that the products are solid spheres with the size around 500 nm.

Figure S1(a) provides the top surface morphology of the TiO_2 NP photoelectrode (see ESI). It can be seen that the surface is distributed with uniform size of ~25 nm TiO_2 NPs, and only small pores around 30 nm exist among NPs. The cross-section view is presented in Fig. S1(b), showing that the thickness of the TiO_2 NP photoelectrode is about 15 μ m with a double-layered structure. Both the top surface and the cross-section SEM images display the homogenous distribution of NPs, indicating a dense TiO_2 NP scaffold film formed on the FTO glass.

When the TiO₂ cavity layer was formed using

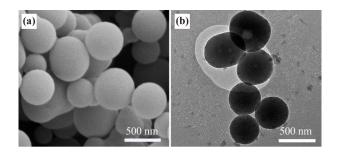


Fig. 1 Morphology of carbonaceous microspheres: (a) SEM image; (b) TEM image.

carbonaceous microspheres as the template, the surface morphology changed obviously comparing with the TiO2 NP photoelectrode. The top morphology SEM image of TiO₂ C/NP (12.5 wt.%) is shown in Fig. 2(a). There are minority of cavities formed among TiO2 NPs in comparison with Fig. S1(a). The size of these cavities is around 350 nm, which is much bigger than pores observed in Fig. S1(a), indicating that the cavities are built by burning out carbonaceous microspheres. noticeable change is that the number of cavities increased as the weight ratio of carbonaceous microspheres was adjusted from 12.5 to 25 and 37.5 wt.%, as shown in Figs. 2(b) and 2(c). This variation demonstrates that the number of cavities can be easily tuned in the doublelayered TiO2 C/NP photoelectrode by adding different quantities of carbonaceous microspheres. Figure 2(d) exhibits the cross-section view of TiO₂ C/NP (25 wt.%). Clearly, a new TiO₂ NP layer embedded with cavities formed on the top of the dense TiO2 NP layer. It can be discerned from Fig. 2(e) that these ~350 nm size of cavities distributed randomly across the top layer of TiO2 C/NP, which is in agreement with Fig. 2(b).

The formation mechanism of TiO₂ cavities layer on top of TiO₂ NPs layer is presented in Fig. 3. As hard template, carbonaceous microspheres play important role in making cavities. During the heating process at 450 °C, carbonaceous microspheres in the mixture containing TiO₂ NPs and carbonaceous microspheres on the top of the TiO₂ NP layer were burned out gradually, leaving cavities surrounded by TiO₂ NPs. Therefore, the numbers of cavities in the top layer can be easily controlled just by varying the weight ratio of carbonaceous microspheres,

forming different types of TiO₂ C/NP photoelectrodes. On the basis of the light scattering theory [31], these cavities may be possible to generate strong light-scattering effect in the double-layered TiO₂ C/NP photoelectrode.

Based on TiO2 C/NP photoelectrodes, DSSCs have been assembled and their J-V performances have been also investigated. Figure 4 presents J-V curves of the DSSCs assembled with the TiO₂ NP photoelectrode and three types of TiO₂ C/NP photoelectrodes. Table 1 also lists the important photovoltaic parameters including short-circuit current density (J_{sc}) , open circuit voltage $(V_{\rm oc})$, fill factor (FF), and power conversion efficiency (PCE). The other two photoelectrodes including TiO₂ C/NP (6.25 wt.%) and TiO_2 C/NP (50 wt.%) have also been tested and their J-V curves are presented in Fig. S3 (see ESI). The results show that the photovoltaic performances of DSSCs based on such two TiO2 C/NP photoelectrodes are lower than that of the DSSC based on any photoelectrodes of TiO₂ C/NP (12.5 wt.%), TiO₂ C/NP (25 wt.%), and TiO₂ C/NP (37.5 wt.%). Therefore, we focused on the investigation of the TiO2 C/NP photoelectrode prepared separately with 12.5, 25, and 37.5 wt.% carbonaceous microspheres. The DSSC based on the TiO2 NP photoelectrode shows the PCE of 4.57% with the $J_{\rm sc}$ of 10.73 mA·cm⁻², the $V_{\rm oc}$ of 0.71 V, and the FF of 0.60, while for the DSSC based on the TiO2 NP photoelectrode on the top of which TiO₂ cavities formed, the photovoltaic performance improved significantly, especially for the $J_{\rm sc}$ and the PCE. For example, the $J_{\rm sc}$ of the DSSC based on TiO2 C/NP (12.5 wt.%) increased to 15.20 mA·cm⁻², with the $V_{\rm oc}$ of 0.72 V and the FF of 0.55, producing the PCE of 6.02%. Moreover, the number

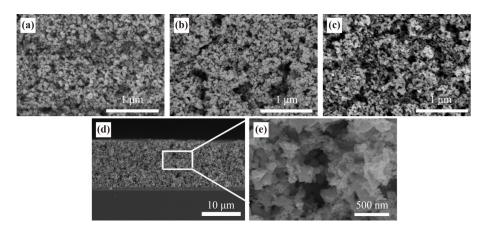


Fig. 2 (a)(b)(c) Top surface morphologies of TiO₂ C/NP photoelectrodes prepared by 12.5, 25, and 37.5 wt.% carbonaceous microspheres (from left to right). (d) The cross-section view of the TiO₂ C/NP photoelectrode. (e) Magnified SEM image of the selected local zone from panel (d).

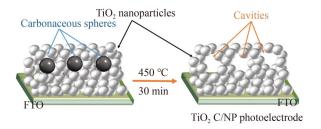


Fig. 3 The formation mechanism of the ${\rm TiO_2}$ C/NP photoelectrode using carbonaceous microspheres as the hard template.

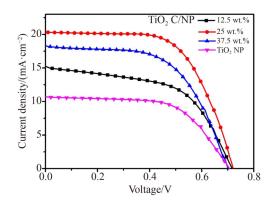


Fig. 4 J–V curves of DSSCs based on TiO₂ NP, TiO₂ C/NP (12.5 wt.%), TiO₂ C/NP (25 wt.%), and TiO₂ C/NP (37.5 wt.%).

Table 1 J–V parameters of DSSCs assembled with different photoelectrodes of TiO₂ NP and three types of TiO₂ C/NP

Photoelectrode	DSSC performance				
	$J_{\rm sc}/({\rm mA\cdot cm^{-2}})$	V _{oc} /V	FF	PCE/%	
TiO ₂ NP	10.73	0.71	0.60	4.57	
TiO ₂ C/NP (12.5 wt.%)	15.20	0.72	0.55	6.02	
TiO ₂ C/NP (25 wt.%)	20.35	0.72	0.62	9.08	
TiO ₂ C/NP (37.5 wt.%)	18.31	0.70	0.58	7.43	

of cavities in TiO_2 C/NP also has influence on the photovoltaic performance. The DSSC based on TiO_2 C/NP (25 wt.%) shows the $J_{\rm sc}$ of 20.35 mA·cm⁻², the $V_{\rm oc}$ of 0.72V, the FF of 0.62, and the PCE of 9.08%, in which the $J_{\rm sc}$ and the PCE are the highest in the sample solar cells. However, a further increase of the cavity number in the TiO_2 C/NP photoelectrode reduced the DSSC performance. The TiO_2 C/NP (37.5 wt.%) produces the PCE of 7.43% with the $J_{\rm sc}$ of 18.31 mA·cm⁻², the $V_{\rm oc}$ of 0.70 V, and the FF of 0.58. On the basis of the J–V curve, it can be found that the great improvement of the $J_{\rm sc}$ is the most key factor that contributes to the increment of the PCE.

The only difference of these DSSCs is the cavities

formed in the top layer of TiO₂ C/NP in comparison with the TiO₂ NP photoelectrode. Therefore, we infer that TiO₂ C/NP may combine advantages of TiO₂ NP and cavities together, resulting in the enhancement of the photovoltaic performance. Figure 5 sketches the configuration of the DSSC based on the TiO₂ C/NP photoelectrode. The TiO₂ C/NP may have two effects: (i) guarantee of almost the same dye adsorption comparing to TiO₂ NP, and (ii) stronger light-scattering effect caused by cavities.

Figure 6 provides N_2 adsorption–desorption curves of TiO_2 C/NP (25 wt.%), showing the specific surface area is 59.18 m²·g⁻¹ which is very close to that of TiO_2 NP as shown in Fig. S2 (see ESI). Both TiO_2 C/NP and TiO_2 NP photoelectrodes are built by P25, so the specific surface area changed slightly. This also proves that the formation of TiO_2 C/NP can continue to maintain the advantages of TiO_2 NPs in the dye adsorption.

It is well known that the light scattering of the TiO₂ photoelectrode is a crucial factor influencing the light harvesting efficiency (LHE) and then the PCE of DSSCs [31]. Figure 7(a) compares the DRS properties of TiO₂ NP

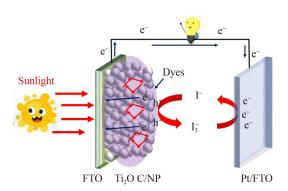


Fig. 5 Configuration of the DSSC based on the ${\rm TiO_2~C/NP}$ photoelectrode.

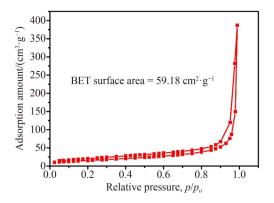


Fig. 6 N₂ adsorption–desorption curves of TiO₂ C/NP (25 wt.%).

and TiO₂ C/NP (25 wt.%). Obviously, TiO₂ C/NP has a higher reflectance than TiO₂ NP in the visible region, indicating a stronger light scattering. In the case of TiO₂ C/NP, the cavities in the size of ~350 nm is beneficial to generate stronger light scattering than TiO₂ NP because the cavity size is close to the visible-light wavelength [32]. A stronger light scattering can improve the LHE. Therefore, it is anticipatable that an increment of IPCE can be achieved with the DSSC based on TiO₂ C/NP. The relationship between IPCE and LHE is as follows [33]:

$$IPCE = LHE \times \Phi_{inj} \times \eta_{cc}$$
 (1)

where $\Phi_{\rm inj}$ and $\eta_{\rm cc}$ represent the charge injection efficiency and the charge collection efficiency, respectively. Figure 7 (b) illustrates the IPCE of DSSCs assembled with TiO₂ NP and TiO₂ C/NP (25 wt.%), demonstrating that TiO₂ C/NP contributes a higher IPCE. This result is in consistence with the $J_{\rm sc}$ in Fig. 4 which is expressed by Eq. (2) [34] as follows:

$$J_{\rm sc} = \int_{\lambda_1}^{\lambda_2} F(\lambda) \delta(\lambda) d\lambda \tag{2}$$

where λ_1 and λ_2 denote the starting point and the ending point of the wavelength, respectively, $F(\lambda)$ represents the photon flux, and δ represent the IPCE. The TiO₂ C/NP (25 wt.%) photoelectrode generates stronger light scattering and boosts the LHE, improving the IPCE and the $J_{\rm Sc}$.

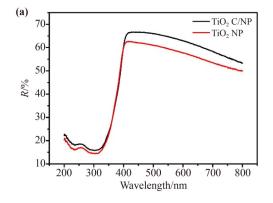
Although the existence of cavities on the top of TiO_2 NP layer plays an important role in promoting J_{sc} , the further increment of the cavity amount such as TiO_2 C/NP (37.5 wt.%) decreased the J_{sc} to 18.31 mA·cm⁻², which indicates that there may be an opposite effect between light scattering and charges recombination in the TiO_2

C/NP photoelectrode. Therefore, we speculate that more cavities in the TiO_2 C/NP photoelectrode may prolong the pathway of electrons, increasing the opportunities of charges recombination. In order to verify this speculation, the OCVD measurement results are shown in Fig. 8(a). Interestingly, the TiO_2 C/NP (25 wt.%) solar cell exhibits a faster V_{oc} decay rate than the DSSC based on TiO_2 NP and a slower decay rate than the DSSC based on TiO_2 C/NP (37.5 wt.%). The electron lifetime (τ_n) can be directly revealed by the slope of the V_{oc} decay curve according to Eq. (3) [35]:

$$\tau_{\rm n} = -\frac{k_{\rm B}T}{e} \left(\frac{\mathrm{d}V_{\rm oc}}{\mathrm{d}t}\right)^{-1} \tag{3}$$

where e, $k_{\rm B}$, and T denote the electronic charge, the Bolzman constant, and the absolute temperature, respectively. The plot of $\tau_{\rm n}$ versus $V_{\rm oc}$ are exhibited in Fig. 8(b), indicating that the order of electron lifetime from long to short is: TiO₂ NP, TiO₂ C/NP (25 wt.%), and TiO₂ C/NP (37.5 wt.%).

To further reveal the charges recombination dynamics, the EIS results of DSSCs assembled with TiO_2 NP, TiO_2 C/NP (25 wt.%), and TiO_2 C/NP (37.5 wt.%) are shown in Fig. 9. The spectra which are fitted by an equivalent circuit (inset of Fig. 9) shows two semi arcs. The small semi arc in high-frequency region is attributed to the recombination resistance at the counter electrode–electrolyte interface (R_{CE}) [36]. Another larger semi arc in low frequency represents the recombination resistance of photoexcited electrons at the TiO_2 photoelectrode–electrolyte interface (R_{rec}) [37]. The fitting results summarized from EIS data which are fitted by "Zview" software are listed in Table 2. A bigger second semi arc in low frequency indicates a higher recombination resistance



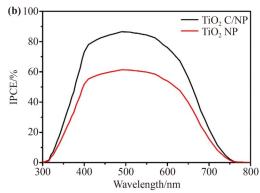
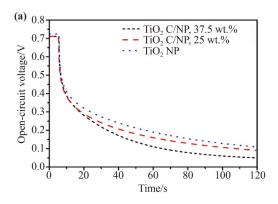


Fig. 7 (a) Diffused reflectance spectra of TiO₂ NP and TiO₂ C/NP (25 wt.%). (b) IPCE of DSSCs assembled with TiO₂ NP and TiO₂ C/NP (25 wt.%).



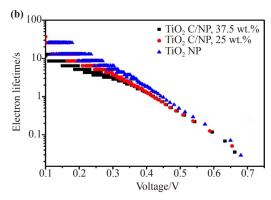


Fig. 8 (a) OCVD curves of DSSCs assembled with TiO_2 NP, TiO_2 C/NP (25 wt.%), and TiO_2 C/NP (37.5 wt.%). (b) Electron lifetime of DSSCs assembled with TiO_2 NP, TiO_2 C/NP (25 wt.%), and TiO_2 C/NP (37.5 wt.%).

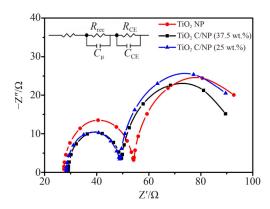


Fig. 9 EIS results of DSSCs assembled with TiO_2 NP, TiO_2 C/NP (25 wt.%), and TiO_2 C/NP (37.5 wt.%). The inset is equivalent circuit of DSSC.

Table 2 EIS fitting data of DSSCs assembled with different photoelectrodes of TiO₂ NP, TiO₂ C/NP (25 wt.%), and TiO₂ C/NP (37.5 wt.%)

Photoelectrode	EIS fitting results of the DSSC				
	R_{CE}/Ω	$R_{\rm rec}/\Omega$	$C_{\mu}/\mu F$	$ au_{ m n}/{ m ms}$	
TiO ₂ NP	27	51	648	32	
TiO ₂ C/NP (25 wt.%)	21	49	602	30	
TiO ₂ C/NP (37.5 wt.%)	21	46	505	23	

and a longer electron lifetime. The τ_n is determined by R_{rec} and C_{μ} as follows [38]:

$$\tau_{\rm n} = R_{\rm rec} \times C_{\rm u} \tag{4}$$

Therefore, the electron lifetime values of DSSCs based on TiO₂ NP, TiO₂ C/NP (25 wt.%), and TiO₂ C/NP (37.5 wt.%) photoelectrodes are calculated to be 32, 30, and 23 ms, respectively. The trend of this result is consistent with that of OCVD, both of which verify that there is an opposite effect between light scattering and charges recombination in TiO₂ C/NP photoelectrodes.

However, if the cavity amount is appropriately controlled in the TiO_2 C/NP photoelectrode, it is possible to make the strong light scattering play the key role in enhancing the J_{sc} significantly, leading to the improvement of the PCE, as revealed by the result of TiO_2 C/NP (25 wt.%).

4 Conclusions

Using carbonaceous microspheres as the template, TiO₂ cavities were built on the top of the TiO₂ NP layer, forming the double-layered TiO₂ C/NP photoelectrode for DSSCs. The cavity amount in the TiO₂ C/NP photoelectrode can be controlled by adjusting the weight ratio of the template. The TiO2 C/NP has two positive effects on the increment of the J_{sc} when used in DSSCs, which include (i) guaranteeing almost the same dye adsorption comparing to TiO2 NP, and (ii) enlarging light harvesting efficiency by generating strong light scattering through cavities. Although there is an opposite effect between the light scattering and the charge recombination in TiO₂ C/NP photoelectrodes, the light scattering can play a major role in enhancing the J_{sc} significantly by controlling the cavity amount appropriately in TiO₂ C/NP. Finally, based on the optimized TiO₂ C/NP (25 wt.%), the DSSC enables the PCE reach 9.08%, which is over 98% higher than the DSSC based on just the TiO2 NP photoelectrode.

Acknowledgements This work was financially supported by the National Natural Science Foundation of China (Grant No. 51862007), the Western Youth Scholars Program of Chinese Academy of Sciences, and the Young Doctor Fund Project of Gansu Provincial Department of Education (Grant No. 2022QB-156).

Electronic supplementary information Supplementary materials can be found in the online version at https://doi.org/10.1007/s11706-023-0638-8.

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