

Exploring the Electronic and Optical Properties of $\text{Cu}(\text{In},\text{Ga})\text{Se}_2$

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3 Computational methods

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- Parameterization of energy bands for CIGS
- Dielectric function for $\text{CuIn}_{0.5}\text{Ga}_{0.5}\text{Se}_2$

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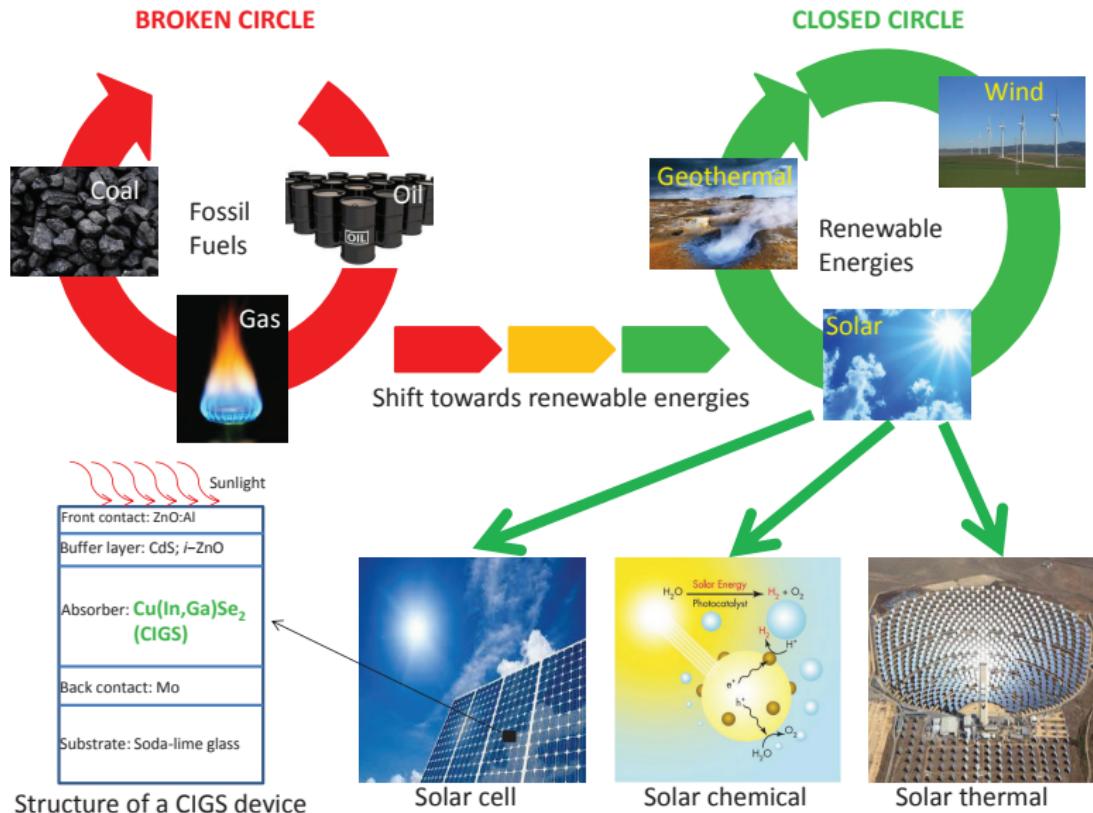
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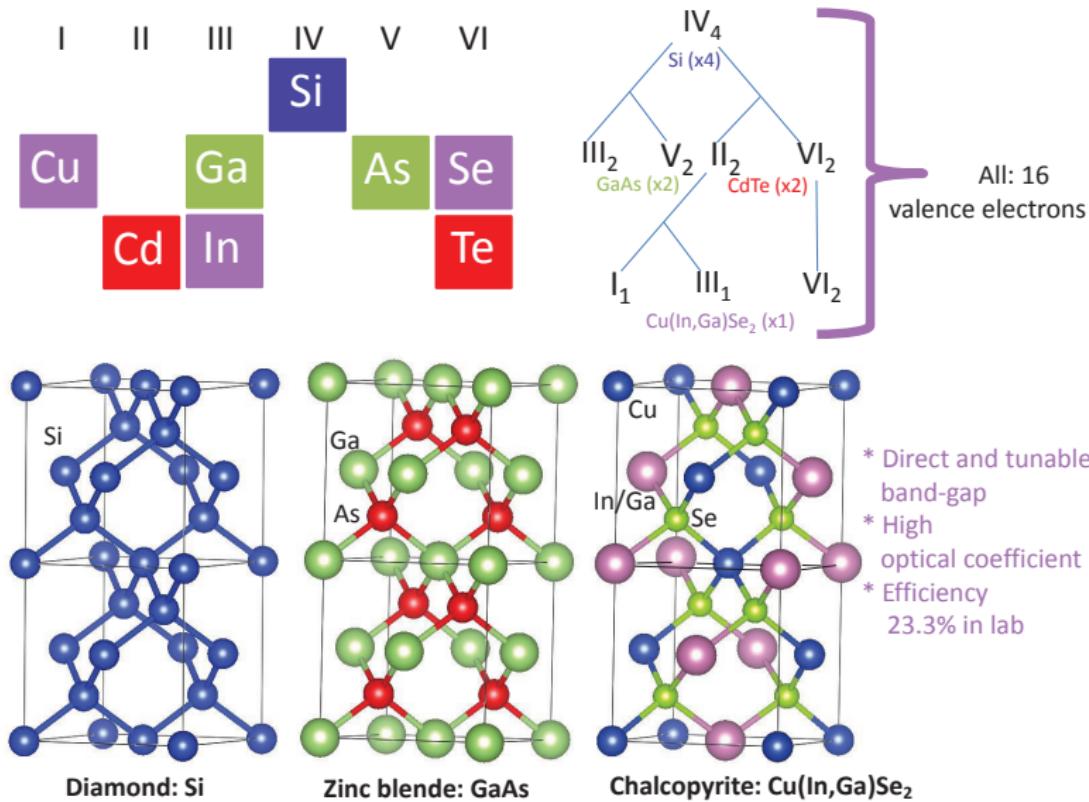
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Which field and material am I working on?



Why Cu(In,Ga)Se₂?



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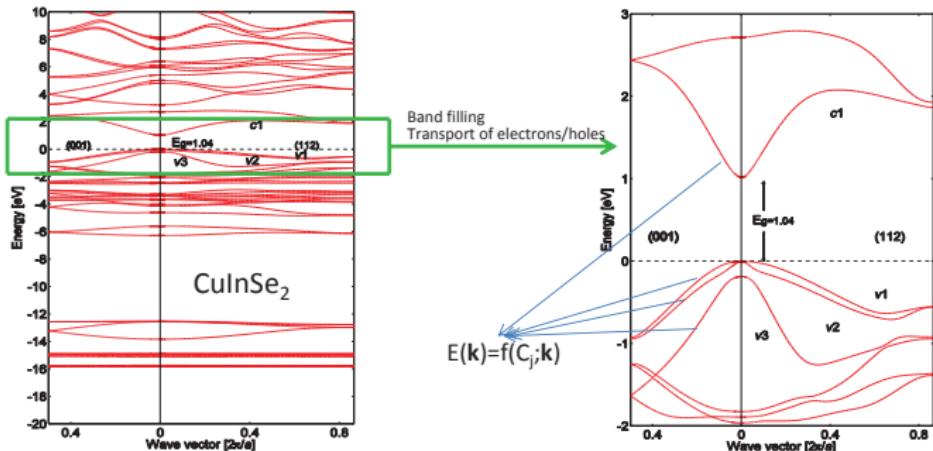
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Which aspects of Cu(In,Ga)Se₂ have I been working on?

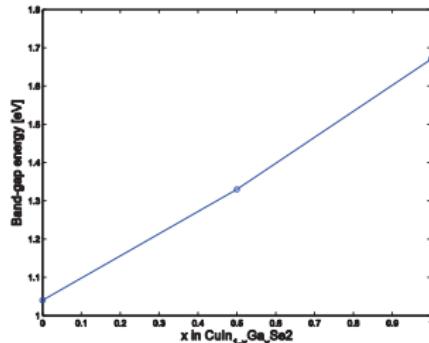
- Energy bands of CuInSe₂, CuIn_{0.5}Ga_{0.5}Se₂, and CuGaSe₂ are calculated by means of theoretical calculations. Parameterization of the lowest conduction (CB) and the three uppermost valence bands (VBs) are explored based on the calculated energy bands. Carrier concentration, Fermi level, and many other aspects are analyzed based on the parameterization.
- Optical properties of CuIn_{0.5}Ga_{0.5}Se₂ are explored by means of theoretical calculations.

Motivation for the parameterization



c1: the lowest conduction band (CB)
v1: the topmost valence band (VB)

Commercially interesting: $\text{CuIn}_{0.7}\text{Ga}_{0.3}\text{Se}_2$,
 $\text{CuIn}_{0.5}\text{Ga}_{0.5}\text{Se}_2$ indicates how CIGS behaves
when alloying In and Ga.



Motivation for calculation of optical properties

- Help experimentalists to understand the dielectric function spectrum.
- Understand details in the optical transition for these types of materials.

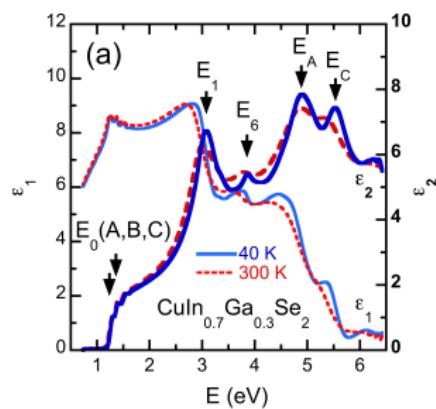


Figure : The real (ϵ_1) and imaginary (ϵ_2) part of dielectric function spectra.

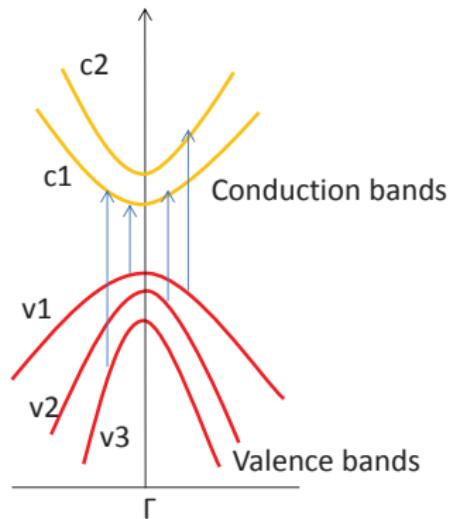


Figure : Energy bands.

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Density functional theory (DFT)

- The first theorem states that the external potential $V_{ext}(\mathbf{r})$ is determined uniquely for any many-electron system by the ground-state electron density ρ .
- The second theorem states that there is a universal functional $F[\rho]$ for the total energy in the terms of the electron density ρ with any external potential $V_{ext}(\mathbf{r})$, and the exact ground-state density is obtained when the ground state total energy functional reaches its minimal value, that is, $E[\rho'] > E[\rho]$. Here, ρ is the exact ground-state density.

Kohn-Sham (KS) Equation

This problem in DFT is solved by KS equation:

$$\begin{aligned} E_{total}[\rho] &= T[\rho] + V_{int}[\rho] + V_{ext}[\rho] \\ &= T_0[\rho] + V_H[\rho] + V_{ext}[\rho] + \underbrace{(T[\rho] - T_0[\rho])}_{(V_{int}[\rho] - V_H[\rho])} \quad (1) \\ &= T_0[\rho] + V_H[\rho] + V_{ext}[\rho] + E_{xc}[\rho]. \end{aligned}$$

The KS equation is derived as

$$\left(-\frac{\nabla^2}{2} + V^{KS}(\mathbf{r}) \right) \Psi_i^{KS}(\mathbf{r}) = \epsilon_i^{KS} \Psi_i^{KS}(\mathbf{r}). \quad (2)$$

The $V^{KS}(\mathbf{r})$ is given as

$$\begin{aligned} V^{KS}(\mathbf{r}) &= V_{ext}(\mathbf{r}) + \int d\mathbf{r}' \frac{\rho(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} + \frac{\delta E_{xc}}{\delta \rho(\mathbf{r})} \\ &= V_{ext}(\mathbf{r}) + \int d\mathbf{r}' \frac{\rho(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} + V_{xc}(\mathbf{r}). \end{aligned} \quad (3)$$

Full-potential linearised augmented-plane wave (FPLAPW)

Unit cell is divided into two regions: one is sphere region called muffin tin (MT) region, which is defined by the center of atom, but non-overlap each sphere; the remaining region is called interstitial (I) region.

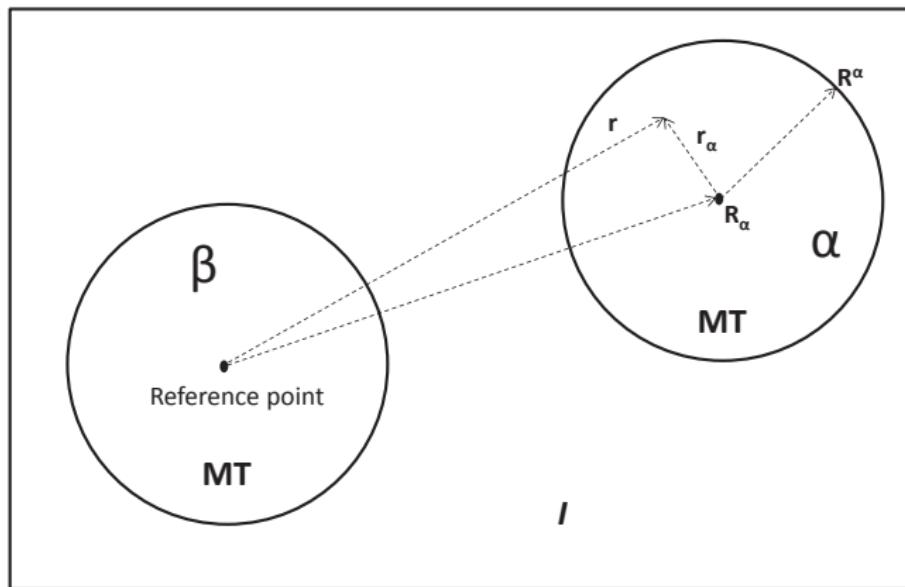


Figure : Partition of the unit cell.

The KS wavefunction can be expanded by a set of basis functions

$$\Psi_{i,\mathbf{k}}^{KS}(\mathbf{r}) = \sum_{\mathbf{G}}^{N_G} C_{i,\mathbf{k}+\mathbf{G}} \phi_{\mathbf{k}+\mathbf{G}}^{LAPW}(\mathbf{r}). \quad (4)$$

Here

$$\phi_{\mathbf{k}+\mathbf{G}}^{LAPW}(\mathbf{r}) = \begin{cases} \frac{1}{\sqrt{\Omega}} e^{i(\mathbf{k}+\mathbf{G})\mathbf{r}} & \text{if } \mathbf{r} \in I \\ \sum_{\alpha} \sum_{\ell m} f_{\ell m}(r_{\alpha}, \mathbf{k} + \mathbf{G}, \epsilon_{\ell, \alpha}) Y_{\ell m}(\hat{\mathbf{r}}_{\alpha}) & \text{if } r_{\alpha} \in MT. \end{cases} \quad (5)$$

The potential in the FPLAPW method is also divided into two regions, the MT region and the I region.

$$V(\mathbf{r}) = \begin{cases} \sum_{\mathbf{G}} V_{\mathbf{G}} e^{i\mathbf{G}\mathbf{r}} & \text{if } \mathbf{r} \in I \\ \sum_{\ell m} V_{\ell m}^{\alpha}(r_{\alpha}) Y_{\ell m}(\hat{\mathbf{r}}_{\alpha}) & \text{if } r_{\alpha} \in MT. \end{cases} \quad (6)$$

Here, $V_{\mathbf{G}}$ and $V_{\ell m}^{\alpha}(r_{\alpha})$ can be decided by electron density.

The process for solving KS equation

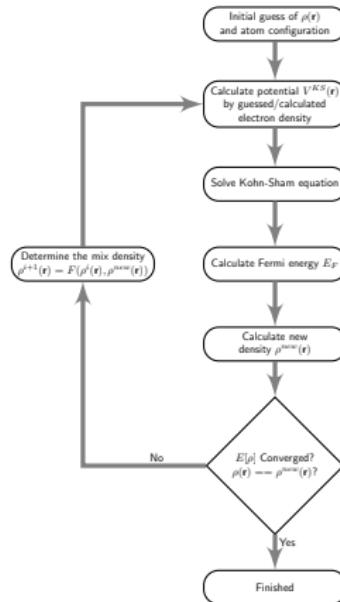


Figure : Flow chart of the $(i + 1)$:th iteration for solving KS equation.

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Result 1: Parabolic band approximation (pba)

$$E_j^{pb}(\mathbf{k}) = E_j(\mathbf{0}) \pm \left[\frac{\tilde{\mathbf{k}}_x^2 + \tilde{\mathbf{k}}_y^2}{m_j^\perp} + \frac{\tilde{\mathbf{k}}_z^2}{m_j^\parallel} \right] \quad (7)$$

$$\tilde{\mathbf{k}}_\alpha^2 = \frac{\hbar^2 \mathbf{k}_\alpha^2}{2e}, \text{ where } \alpha = x, y, \text{ and } z.$$

Result 1: Full band parameterization (fbp)

$$\begin{aligned}
 E_j(\mathbf{k}) = & E_j^{pb}(\mathbf{k}) + E_j^0 + \Delta_{j,1} \left(\delta_{j,1}^2 \left(\frac{\tilde{\mathbf{k}}_x^4 + \tilde{\mathbf{k}}_y^4}{m_0^2} \right) + \delta_{j,2}^2 \left(\frac{\tilde{\mathbf{k}}_x^2 \tilde{\mathbf{k}}_y^2}{m_0^2} \right) + 1 \right)^{1/2} \\
 & + \Delta_{j,2} \left(\delta_{j,3}^3 \left(\frac{\tilde{\mathbf{k}}_x^6 + \tilde{\mathbf{k}}_y^6}{m_0^3} \right) + \delta_{j,4}^3 \left(\frac{\tilde{\mathbf{k}}_x^2 \tilde{\mathbf{k}}_y^4 + \tilde{\mathbf{k}}_x^4 \tilde{\mathbf{k}}_y^2}{m_0^3} \right) + 1 \right)^{1/3} \\
 & + \Delta_{j,3} \left(\delta_{j,5}^2 \left(\frac{\tilde{\mathbf{k}}_z^4}{m_0^2} \right) + 1 \right)^{1/2} + \Delta_{j,4} \left(\delta_{j,6}^3 \left(\frac{\tilde{\mathbf{k}}_z^6}{m_0^3} \right) + 1 \right)^{1/3} \\
 & + \Delta_{j,5} \left(\delta_{j,7}^2 \left(\frac{\tilde{\mathbf{k}}_x^2 \tilde{\mathbf{k}}_z^2 + \tilde{\mathbf{k}}_y^2 \tilde{\mathbf{k}}_z^2}{m_0^2} \right) + 1 \right)^{1/2} \\
 & + \Delta_{j,6} \left(\delta_{j,8}^3 \left(\frac{\tilde{\mathbf{k}}_x^4 \tilde{\mathbf{k}}_z^2 + \tilde{\mathbf{k}}_y^4 \tilde{\mathbf{k}}_z^2}{m_0^3} \right) + \delta_{j,9}^3 \left(\frac{\tilde{\mathbf{k}}_x^2 \tilde{\mathbf{k}}_z^4 + \tilde{\mathbf{k}}_y^2 \tilde{\mathbf{k}}_z^4}{m_0^3} \right) + \delta_{j,10}^3 \left(\frac{\tilde{\mathbf{k}}_x^2 \tilde{\mathbf{k}}_y^2 \tilde{\mathbf{k}}_z^2}{m_0^3} \right) + 1 \right)^{1/3}
 \end{aligned} \tag{8}$$

Result 1: Parameterization of band structure for CIGS

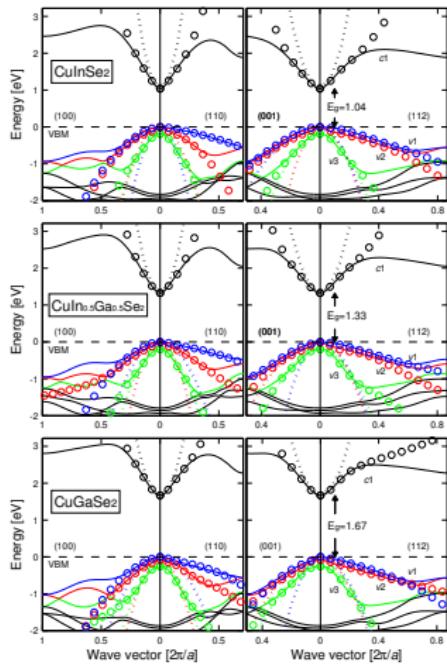


Figure : Electronic band structure along four directions. The circles are the results of the fbp, and the dotted lines represent the pba.

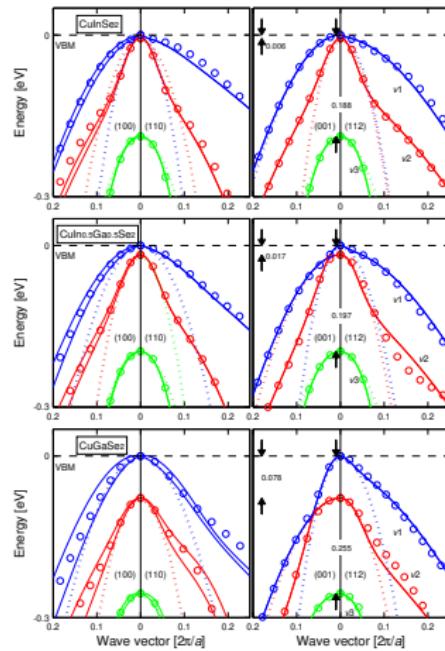


Figure : Close-up of right figure.

Result 1: Non-parabolicity of the energy bands

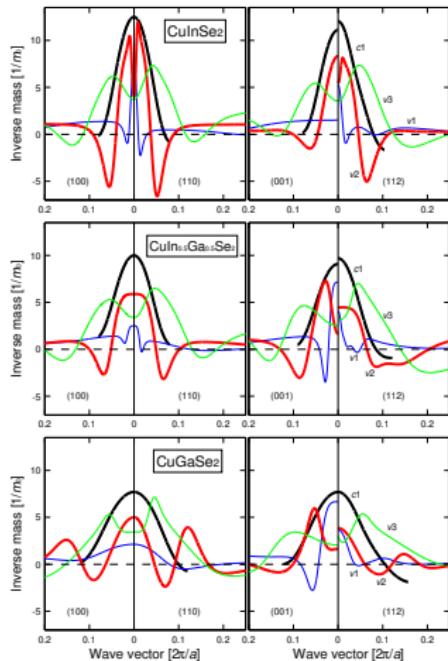


Figure : Inverse of the effective electron and hole masses in the four symmetry directions for the CuIn_{1-x}Ga_xSe₂ ($x = 0, 0.5$, and 1).

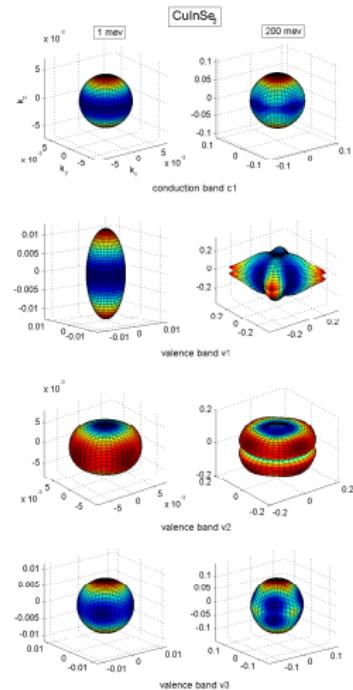


Figure : Constant energy surfaces for the lowest CB and three uppermost VBs in CuInSe₂.

Result 1: Density-of-states (DOS)

In the case of the pba, the density-of-states can be written as

$$g_j^{pba}(E) = \frac{1}{2\pi^2} \left(\frac{2m_j^{DOS}}{\hbar^2} \right)^{3/2} \sqrt{|E - E_j(\mathbf{0})|}. \quad (9)$$

Here, the DOS mass m_j^{DOS} is equal to $(m_j^\perp m_j^\perp m_j^{\parallel})^{1/3}$, which represents the extent of filling the specific band with free carriers to certain energy. In order to take advantage of the simple **Eq. 9** for the non-parabolic energy bands, the energy-dependent DOS mass ($m_{v/c}^{DOS}$), which contains the non-parabolicity and anisotropy of the band dispersion, is defined as

$$g_{v/c}(E) = \sum_j g_j(E) = \frac{1}{2\pi^2} \left(\frac{2m_{v/c}^{DOS}(E)}{\hbar^2} \right)^{3/2} \sqrt{|E - E_{v1/c1}(\mathbf{0})|}. \quad (10)$$

Result 1: Density-of-states (DOS) and DOS mass

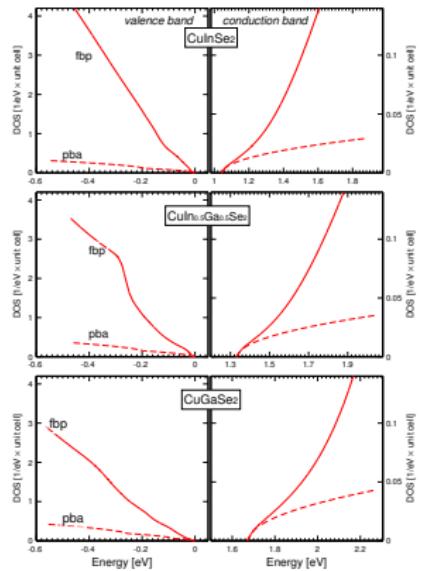


Figure : Total density-of-states of the VBs and CB. The solid lines show the results based on the fbp, and the dashed lines represent the results based on the pba.

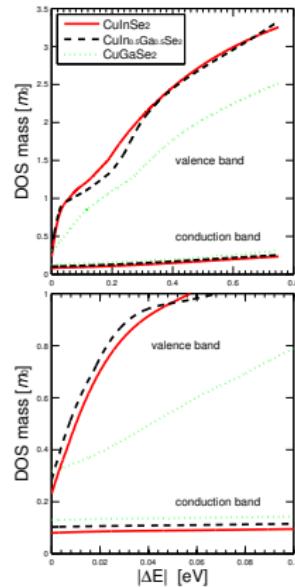


Figure : The DOS mass $m_{v/c}^{DOS}$ is calculated from Eq. 10.

Result 1: Band-gap, Fermi energy and intrinsic carrier concentration

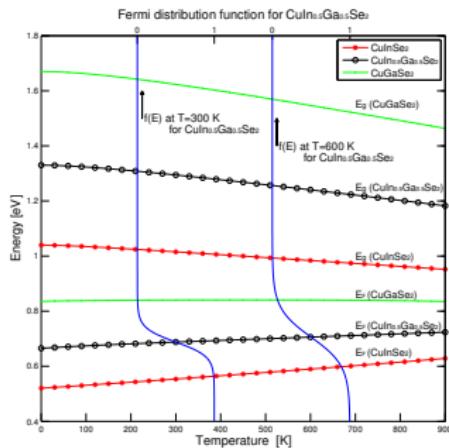


Figure : Band-gap energy and Fermi energy for $1 \leq T \leq 900$ K of intrinsic CuInSe_2 , $\text{CuIn}_{0.5}\text{Ga}_{0.5}\text{Se}_2$, and CuGaSe_2 , determined from the fbp. The Fermi distribution $f(E)$ of $\text{CuIn}_{0.5}\text{Ga}_{0.5}\text{Se}_2$ is presented for $T = 300$ K and 600 K.

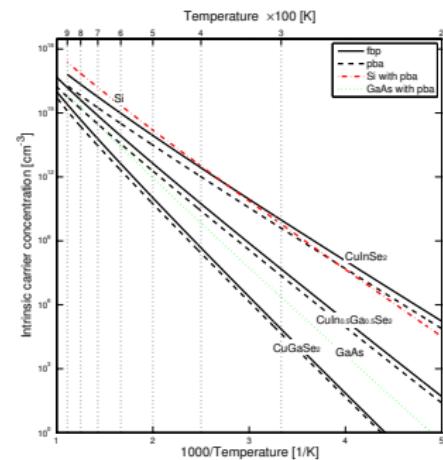


Figure : Intrinsic carrier concentration as function of temperature. For comparison, the theoretical results for GaAs and Si using the pba are given.

Result 1: Carrier concentration in *p*-type for CIGS

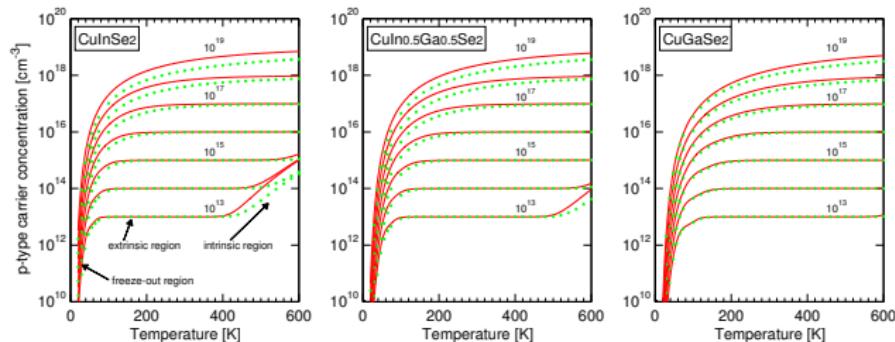


Figure : Free carrier concentration as function of the temperature in *p*-type for CuInSe₂, CuIn_{0.5}Ga_{0.5}Se₂, and CuGaSe₂. The effective doping concentration $N_A = 10^{13}, 10^{14}, 10^{15}, \dots$, and 10^{19} acceptors cm⁻³ are considered.

Result 2: Dielectric function for $\text{CuIn}_{0.5}\text{Ga}_{0.5}\text{Se}_2$

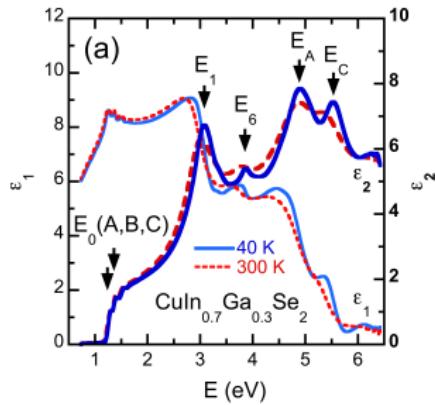


Figure : The real (ϵ_1) and imaginary (ϵ_2) part of dielectric function spectra for $\text{CuIn}_{0.7}\text{Ga}_{0.3}\text{Se}_2$ at 40 K (solid blue line) and 300 K (dashed red lines). Four prominent CP features are shown.

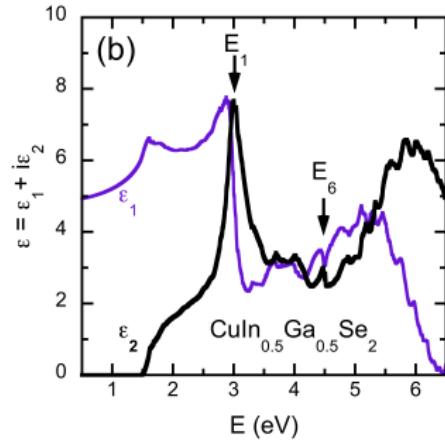


Figure : The dielectric function spectra for $\text{CuIn}_{0.5}\text{Ga}_{0.5}\text{Se}_2$ calculated by FPLAPW method at 0 K. The major CP features are identified.

Result 2: Analysis of the imaginary part of dielectric function

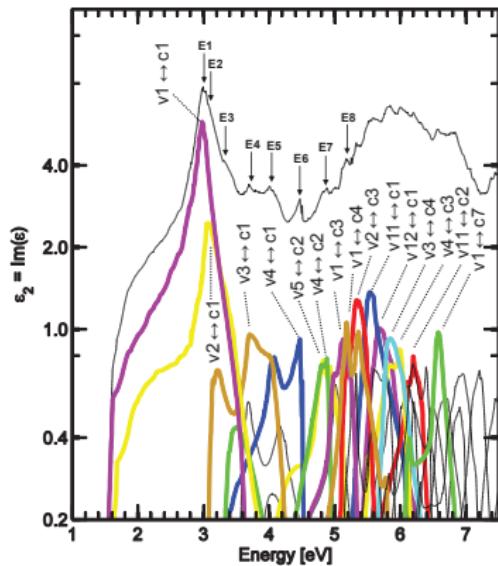


Figure : Band-to-band analysis of the contribution to the total ϵ_2 spectrum. The vertical axis is in the log scale.

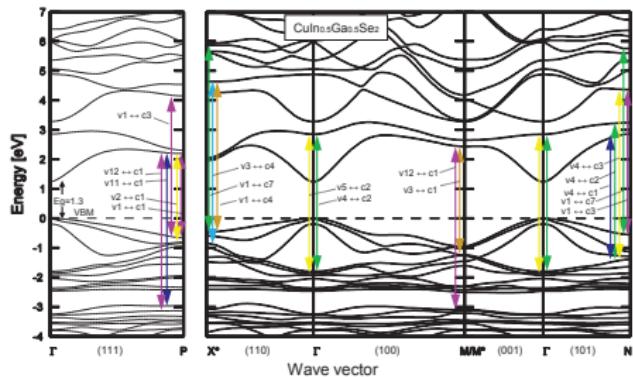


Figure : The calculated electronic band structure of $\text{Culn}_{0.5}\text{Ga}_{0.5}\text{Se}_2$ where the CPs are identified along the main symmetry directions.

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Summary

In this licentiate thesis, two major researches are presented:

- Analysis of the electronic structure of $\text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2$ with $x = 0, 0.5$, and 1. Here, we parameterize the energy bands in order to better describe energy dispersions and better analyze the electron and hole dynamics in the materials. We consider intrinsic and *p*-type materials, and we model the temperature dependence.
- Analysis of the optical properties of $\text{CuIn}_{0.5}\text{Ga}_{0.5}\text{Se}_2$. Here, the dielectric function spectra is calculated and compared with experimental result. The probable electronic origins of the critical point features are discussed as well.

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Future perspectives

- The cost and scarcity of indium in the CIGS device is a problem, and copper zinc tin selenide/sulfide (CZT(S,Se)) can therefore be alternative to CIGS due to the low cost and non-toxicity elements.

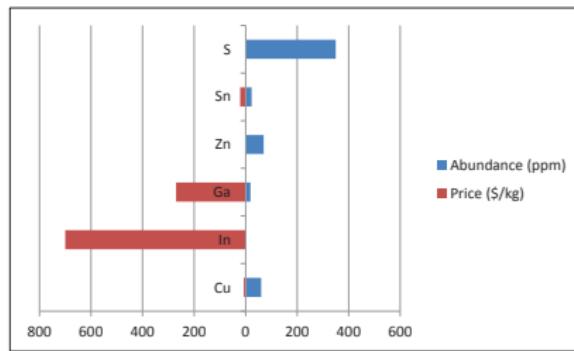


Figure : Abundance and price of Cu, In, Ga, S, Sn, and Zn in Earth's crust (parts per million (ppm) in mass; 10,000 ppm = 1%).

- If there are other similar materials that are of interest. CIGS and CZT(S,Se) can be considered to be in a class denoted as Cu-XY-chalcogenide, where X and Y are two cation elements that replace the group-III In or Ga in CIGS.

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Thank you for your attention!

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