

# New transparent conducting ZnO–In<sub>2</sub>O<sub>3</sub>–SnO<sub>2</sub> thin films prepared by magnetron sputtering

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## Abstract

New multicomponent ZnO–In<sub>2</sub>O<sub>3</sub>–SnO<sub>2</sub> system and new In<sub>4</sub>Sn<sub>3</sub>O<sub>12</sub> transparent conducting oxide thin films have been prepared by RF magnetron sputtering. The In<sub>4</sub>Sn<sub>3</sub>O<sub>12</sub> films, or In<sub>2</sub>O<sub>3</sub>–SnO<sub>2</sub> films with a Sn/(In + Sn) atomic ratio around 0.5, showed a resistivity of  $3\text{--}4 \times 10^{-4} \Omega \text{ cm}$  and an average transmittance above 80% in the visible range when they were prepared at substrate temperatures of room temperature to 350°C. In addition, the electrical properties of multicomponent Zn<sub>2</sub>In<sub>2</sub>O<sub>5</sub>–ZnSnO<sub>3</sub> films changed monotonically as the ZnSnO<sub>3</sub> content was varied. © 1998 Elsevier Science S.A.

**Keywords:** Thin films; Magnetron sputtering; Oxide

## 1. Introduction

Transparent and conductive indium–tin oxide (ITO), films are widely used as transparent electrodes for flat panel displays. Many of the ITO films for practical use have been prepared by magnetron sputtering using an oxide target composed of In<sub>2</sub>O<sub>3</sub> and SnO<sub>2</sub>: 5 to 10 wt.% SnO<sub>2</sub> in content. As a result, studies have been focused on ITO materials consisting of In<sub>2</sub>O<sub>3</sub> doped with less than about 10 wt.% SnO<sub>2</sub> [1].

On the other hand, multicomponent oxides composed of binary and/or ternary compounds have recently attracted much attention as new materials for transparent conducting films [2–4]. As a result of changes in physical properties brought about by controlling the composition of materials in the binary or ternary compounds, these multicomponent transparent conducting oxide thin films may exhibit properties which are suitable for specialized applications. In addition, ternary compounds may possibly improve attainable properties [5–11]. However, thin films of multicomponent oxides composed of In<sub>2</sub>O<sub>3</sub> and SnO<sub>2</sub>, In<sub>2</sub>O<sub>3</sub>–SnO<sub>2</sub> system thin films, have been not reported.

This paper describes the preparation of highly transparent and conductive multicomponent oxide films by RF magnetron sputtering using ZnO–In<sub>2</sub>O<sub>3</sub>–SnO<sub>2</sub> targets. A new transparent conducting In<sub>4</sub>Sn<sub>3</sub>O<sub>12</sub> film is demon-

strated. In addition, transparent conducting Zn<sub>2</sub>In<sub>2</sub>O<sub>5</sub>–ZnSnO<sub>3</sub> system thin films have also been prepared.

## 2. Experimental

Films were prepared by conventional RF planar magnetron sputtering using powder targets. A mixture of In<sub>2</sub>O<sub>3</sub> (purity, 99.99%) and SnO<sub>2</sub> (purity, 99.99%) powders calcined at 1000°C in an argon (Ar) atmosphere for 5 h was used as the target: stainless steel holder, diameter of 80 mm. Substrates of Corning 7059 glass were placed parallel to the target surface at a distance of 35 mm. Sputtering deposition was carried out at sputter gas pressures of 0.1 to 1 Pa in a pure Ar gas or a mixture of Ar and oxygen (O<sub>2</sub>) gases with a RF power of 40 W. The O<sub>2</sub> gas content in the Ar + O<sub>2</sub> gas atmosphere was varied from an O<sub>2</sub> partial pressure of 0 to 8%. Substrate temperatures were varied from room temperature (RT) to 350°C. Although the substrate at RT was not intentionally heated, the surface temperature reached about 180°C after a sputter deposition of 30 min. The deposition rate varied from 10 nm/min for a In<sub>2</sub>O<sub>3</sub> target to 15 nm/min for a SnO<sub>2</sub> target; it was about 11 nm/min for a target with a Sn content (Sn/(In + Sn) atomic ratio) of 0.5. Film thickness was measured using a conventional surface roughness detector with stylus. Electrical resistivity and Hall mobility were measured using the van der Pauw method. Optical transmission through the film was measured in the visible wavelength

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range, 300 to 800 nm. The composition of deposited films was measured by electron probe microanalysis (EPMA).

### 3. Results and discussion

#### 3.1. Preparation of $\text{In}_2\text{O}_3\text{--SnO}_2$ thin films

When  $\text{In}_2\text{O}_3\text{--SnO}_2$  films were prepared by RF magnetron sputtering using  $\text{In}_2\text{O}_3\text{--SnO}_2$  targets with a Sn content of 0 to 1.0, two peaks of carrier concentration were observed depending on the Sn content. The electrical properties as functions of the Sn content are shown in Fig. 1 for  $\text{In}_2\text{O}_3\text{--SnO}_2$  films prepared on substrates at RT and 350°C. The sputter deposition was carried out at a sputter gas pressure of 0.25 Pa in an Ar + O<sub>2</sub> (2%) gas atmosphere. It is well known that the first peak of carrier concentration in films prepared with a Sn content around 0.05 results from the doping effect of Sn into  $\text{In}_2\text{O}_3$ , i.e., ITO [1]. In addition to the maximum carrier concentration, films prepared with a Sn content around 0.05 exhibited the minimum resistivity. The second peak was obtained in films prepared using a target with a Sn content around 0.5, regardless of the substrate temperature. In contrast, the second minimum resistivity of films prepared at RT was obtained using a target with a Sn content of 0.5, but that of films prepared at 350°C was obtained with a Sn content of 0.6.

The obtainable resistivity was dependent on the O<sub>2</sub> gas content introduced into the Ar sputter gas. The electrical properties as functions of the O<sub>2</sub> gas content are shown in Fig. 2 for films prepared on substrates at RT and 350°C using a target with a Sn content of 0.5. The optimal O<sub>2</sub> gas content was about 2%, regardless of the substrate temperature. Films at the second peak exhibited resistivities and carrier concentrations comparable to those of ITO films at

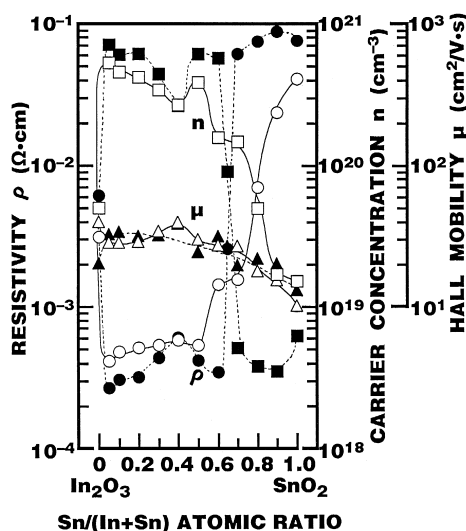


Fig. 1. Resistivity (○, ●), Hall mobility (△, ▲) and carrier concentration (□, ■) as functions of Sn content for  $\text{In}_2\text{O}_3\text{--SnO}_2$  films prepared at RT (open) and 350°C (solid).

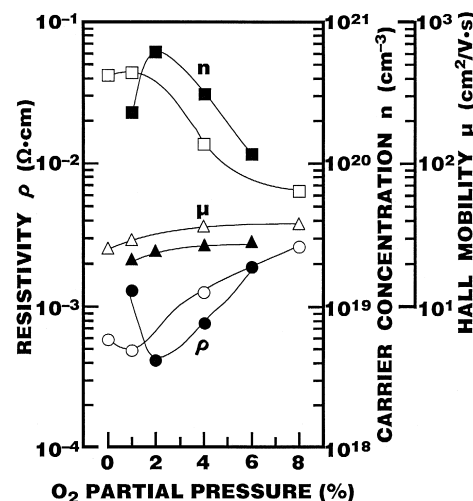


Fig. 2. Resistivity (○, ●), Hall mobility (△, ▲) and carrier concentration (□, ■) as functions of O<sub>2</sub> gas content for films prepared at RT (solid) and 350°C (open) with a Sn content of 0.5.

the first peak. A minimum resistivity of  $3.5 \times 10^{-4} \Omega \text{ cm}$  was obtained for a film prepared at a substrate temperature of 350°C.

#### 3.2. Preparation of $\text{In}_4\text{Sn}_3\text{O}_{12}$ films

From an EPMA of the prepared films, it was found that the Sn content in the films was approximately equal to that present in the target. In addition, the fact that films prepared using targets with a Sn content around 0.5 exhibited a peak of carrier concentration and a minimum of resistivity, as mentioned above, may suggest the existence of a ternary compound composed of  $\text{In}_2\text{O}_3$  and  $\text{SnO}_2$ . The films prepared using targets with a Sn content around 0.5 were identified as  $\text{In}_4\text{Sn}_3\text{O}_{12}$  [12] by the X-ray diffraction analysis shown in Fig. 3, whereas the powder targets with

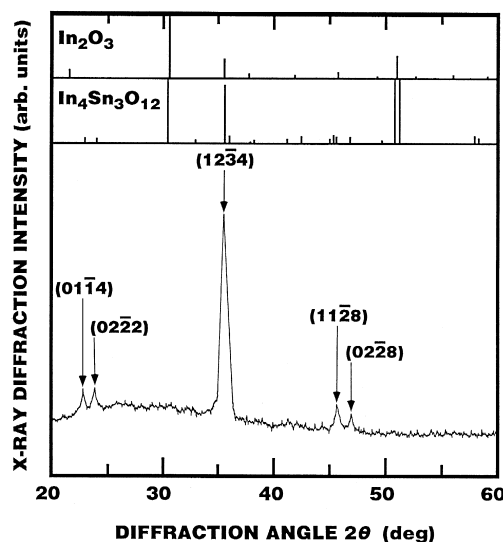


Fig. 3. X-ray diffraction profile for film prepared with a Sn content of 0.5.

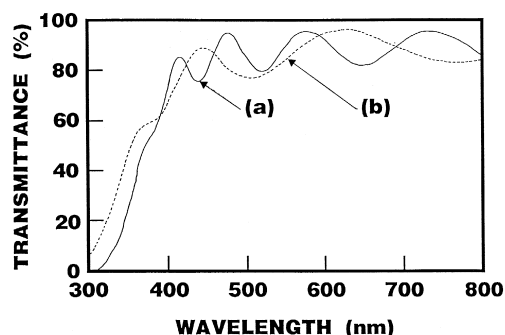


Fig. 4. Optical transmission spectra for  $\text{In}_4\text{Sn}_3\text{O}_{12}$  films prepared at RT (a) and 350°C (b).

a Sn content around 0.5, calcined at 1000°C, were a  $\text{In}_2\text{O}_3$  and  $\text{SnO}_2$  powder mixture. Enoki and Echigoya [12] have recently reported  $\text{In}_4\text{Sn}_3\text{O}_{12}$  as a new material in the  $\text{In}_2\text{O}_3$ – $\text{SnO}_2$  system;  $\text{In}_4\text{Sn}_3\text{O}_{12}$  was identified from an electron diffraction analysis of powder composed of  $2\text{In}_2\text{O}_3$  and  $3\text{SnO}_2$ , annealed at 1227°C. Although the X-ray diffraction profile from  $\text{In}_4\text{Sn}_3\text{O}_{12}$  is very close in that from  $\text{In}_2\text{O}_3$ , as seen from the ASTM<sup>1</sup> and the Enoki and Echigoya data indicated in Fig. 3, there is a clear difference at  $2\theta$  between 20° and 25°: the diffraction profile from  $\text{In}_4\text{Sn}_3\text{O}_{12}$  shows two peaks, but that from  $\text{In}_2\text{O}_3$  is single. As can be seen in Fig. 3, the diffraction profile from films deposited at RT with an  $\text{O}_2$  gas content of 2% using targets with a Sn content of 0.5 exhibited two peaks. It should be noted that the X-ray diffraction profile was strongly dependent on preparation conditions such as the  $\text{O}_2$  gas content, the sputter gas pressure and the substrate temperature. In addition to the two peaks in the X-ray diffraction profile, the angle of the observed diffraction peaks coincided well to the  $\text{In}_4\text{Sn}_3\text{O}_{12}$  data [12]. Therefore, it can be concluded that a polycrystalline  $\text{In}_4\text{Sn}_3\text{O}_{12}$  film with a resistivity as low as  $3 \times 10^{-4} \Omega \text{ cm}$  was prepared by RF magnetron sputtering using targets with a Sn content around 0.5.

Fig. 4 shows optical transmission spectra for  $\text{In}_4\text{Sn}_3\text{O}_{12}$  films prepared at a substrate temperature of RT and 350°C. The  $\text{In}_4\text{Sn}_3\text{O}_{12}$  films were prepared at a sputter gas pressure of 0.25 Pa with an  $\text{O}_2$  gas content of 2% using targets with a Sn content of 0.5. The thicknesses of films prepared at RT and 350°C were 580 and 330 nm, respectively. An average transmittance above 80% in the visible range was obtained for these  $\text{In}_4\text{Sn}_3\text{O}_{12}$  films. The band-gap energy is roughly estimated to be about 3.5 eV and the refractive index to be about 2.1 for  $\text{In}_4\text{Sn}_3\text{O}_{12}$  films [10].

### 3.3. Properties of multicomponent $\text{Zn}_2\text{In}_2\text{O}_5$ – $\text{ZnSnO}_3$ films

We have reported previously that highly transparent and conductive films were realized by  $\text{Zn}_2\text{In}_2\text{O}_5$  [10] and

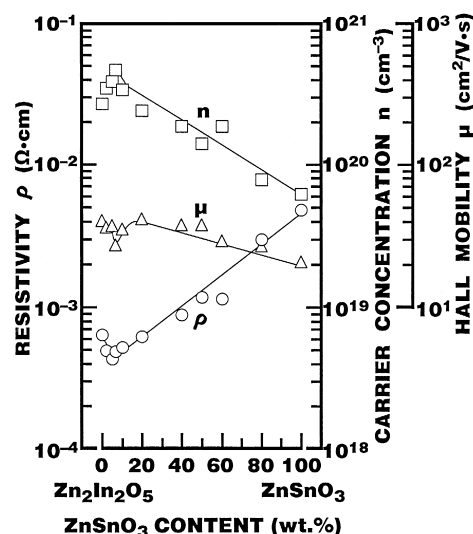


Fig. 5. Resistivity (○), Hall mobility (Δ) and carrier concentration (□) as functions of  $\text{ZnSnO}_3$  content for  $\text{Zn}_2\text{In}_2\text{O}_5$ – $\text{ZnSnO}_3$  films prepared at RT.

$\text{ZnSnO}_3$  [7], both ternary compounds composed of  $\text{ZnO}$  and  $\text{In}_2\text{O}_3$  or  $\text{SnO}_2$ . Now, multicomponent  $\text{Zn}_2\text{In}_2\text{O}_5$ – $\text{ZnSnO}_3$  films have been prepared using  $\text{ZnO}$ – $\text{In}_2\text{O}_3$ – $\text{SnO}_2$  targets. The electrical properties as functions of the  $\text{ZnSnO}_3$  content are shown in Fig. 5 for multicomponent  $\text{Zn}_2\text{In}_2\text{O}_5$ – $\text{ZnSnO}_3$  films prepared on substrates at RT. The sputter deposition was carried out at a sputter gas pressure of 1.2 Pa in a pure Ar atmosphere. It was found that the electrical, optical and chemical properties of the  $\text{Zn}_2\text{In}_2\text{O}_5$ – $\text{ZnSnO}_3$  films changed monotonically as the  $\text{ZnSnO}_3$  content was varied. In  $\text{In}_2\text{O}_3$ – $\text{SnO}_2$  system films, the etching rate in acid solutions such as HCl decreased as the Sn content was increased. In contrast, the etching rate of  $\text{Zn}_2\text{In}_2\text{O}_5$ – $\text{ZnSnO}_3$  system films was improved by the introduction of Zn. The etching rate of  $\text{Zn}_2\text{In}_2\text{O}_5$ – $\text{ZnSnO}_3$  system films in HCl solutions decreased as the  $\text{ZnSnO}_3$  content was increased; the films prepared with a  $\text{ZnSnO}_3$  content above 30 at% were hardly etched in 0.2 M HCl solution.

## 4. Conclusions

Transparent conducting multicomponent oxide thin films have been prepared by RF magnetron sputtering using  $\text{ZnO}$ – $\text{In}_2\text{O}_3$ – $\text{SnO}_2$  targets. When films were prepared using  $\text{In}_2\text{O}_3$ – $\text{SnO}_2$  targets with a Sn content (Sn/(In + Sn) atomic ratio) of 0 to 1.0, two peaks of carrier concentration were observed depending on the Sn content.  $\text{In}_2\text{O}_3$ – $\text{SnO}_2$  films prepared with a Sn/(In + Sn) atomic ratio around 0.5 were identified as  $\text{In}_4\text{Sn}_3\text{O}_{12}$ . A resistivity of  $3\text{--}4 \times 10^{-4} \Omega \text{ cm}$  and an average transmittance above 80% in the visible range were obtained in  $\text{In}_4\text{Sn}_3\text{O}_{12}$  films prepared at substrate temperatures of room temperature to 350°C. In addition, multicomponent  $\text{Zn}_2\text{In}_2\text{O}_5$ – $\text{ZnSnO}_3$  films were prepared using  $\text{ZnO}$ – $\text{In}_2\text{O}_3$ – $\text{SnO}_2$  targets. The

<sup>1</sup> ASTM X-ray Powder Diffraction File, Inorganic, No. 6-0416.

electrical properties of the multicomponent oxide films changed monotonically as the  $\text{ZnSnO}_3$  content was varied. This new material,  $\text{In}_4\text{Sn}_3\text{O}_{12}$ , is promising as a transparent conducting oxide film because of its low cost in comparison with ITO films with a high In content.

### Acknowledgements

The authors wish to acknowledge Y. Murota, T. Kishino and H. Yoshino for their technical assistance in the experiments. This work was partially supported by a Grant-in-Aid for Scientific Research No. 08555079 from the Ministry of Education, Science and Culture of Japan.

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