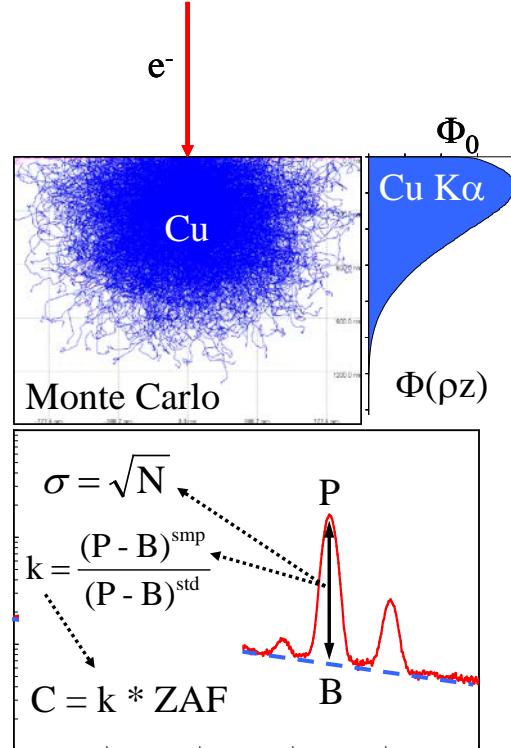


Electron-Probe Microanalysis (EPMA): Summary for Microprobe Lab Demo

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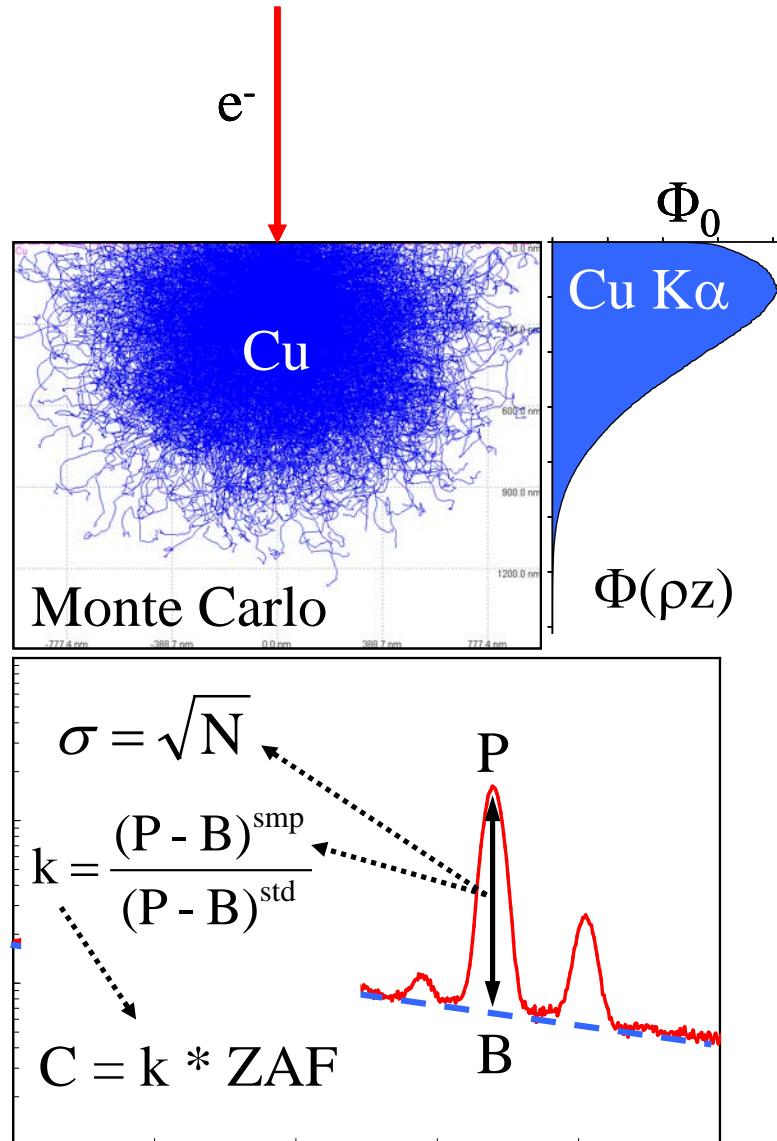
EPMA Summary

Electron-Probe Microanalysis: Summary

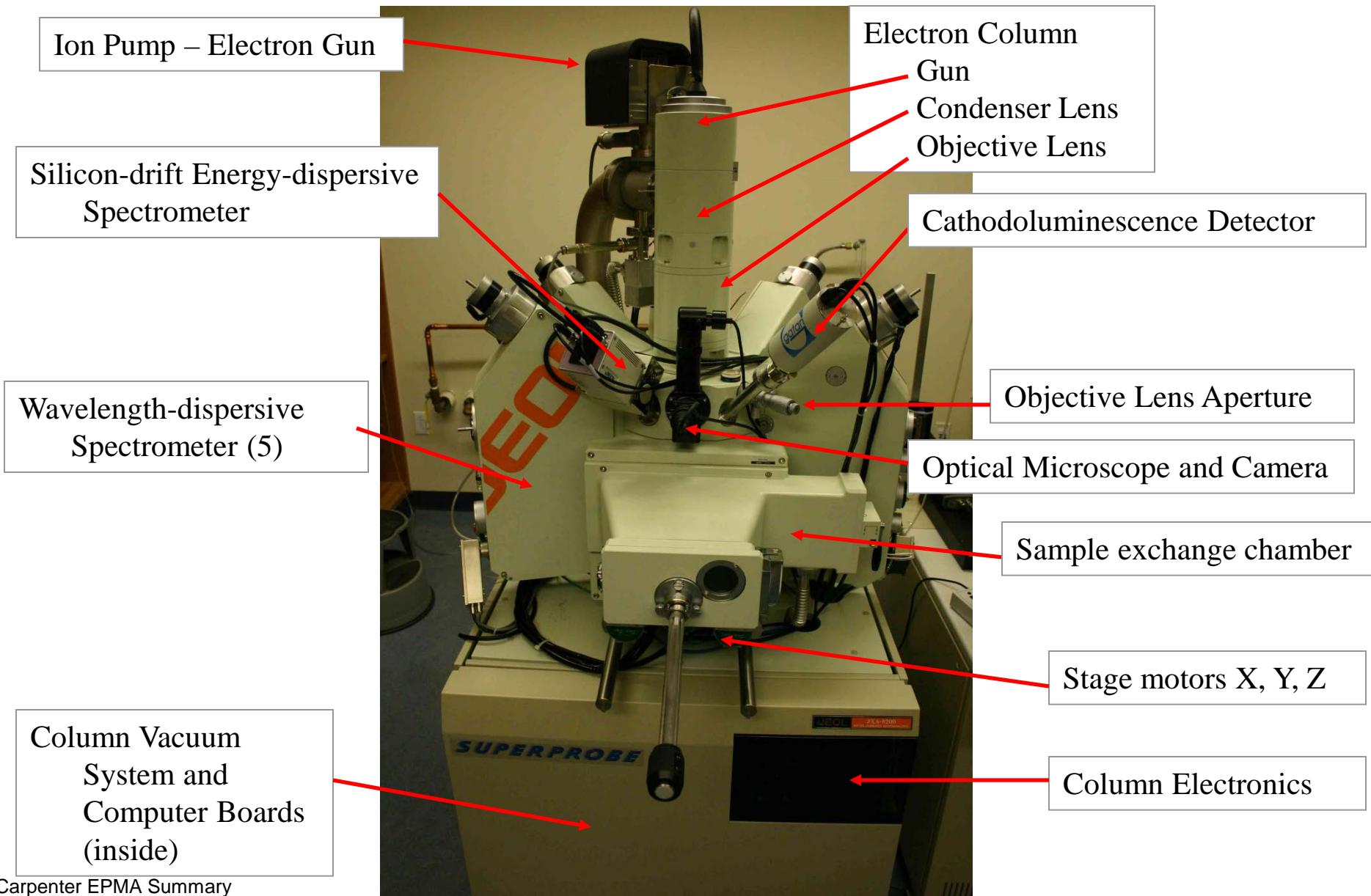
- ◆ EPMA is a non-destructive comparative method where a focused electron beam excites characteristic X-rays from a ~ 1 micron sample volume, and the emitted intensities are converted to concentration units
- ◆ Sample: polished, electrically conductive, inorganic materials
- ◆ Focused electron beam ~ 5 nm, excitation volume ~ 1 micron
- ◆ Electron scattering and energy-loss defines analytical volume
- ◆ Comparison method
 $k\text{-ratio} = (P-B)^{\text{smp}} / (P-B)^{\text{std}}$, background corrected peak intensity
Relative to known standard: $k=0.5$ for 500cps smp vs. 1000 cps std
- ◆ Correction for matrix effects: ZAF multiplicative factors
 - A X-ray absorption
 - Z electron scattering
 - F X-ray fluorescence
- ◆ Concentration calculated from $C = k * \text{ZAF}$, iteratively

EPMA Schematic Summary

- ◆ Sample scattering volume important
- ◆ Depth distribution of X-rays $\Phi(\rho z)$ curve – area = total intensity
- ◆ Correction requires both generated and emitted intensity distribution
- ◆ Precision defined by N counts standard deviation: $\text{sqrt}(N)$
- ◆ Concentration depends on standards and correction algorithm physics
- ◆ Accuracy 2-5% relative
- ◆ Elements Be – U analyzed
- ◆ Wavelength and energy-dispersive spectrometry methods
- ◆ Discrete point analysis vs. map area “wide area analysis”



Electron Microprobe at Washington University



Summary of EPMA Procedure

- ◆ Sample prep and carbon coating
- ◆ Sample imaging and energy-dispersive analysis performed to generate base map and element list.
- ◆ Microprobe stabilized, software setup, elements and standards selected.
- ◆ Wavelength spectrometers peaked on standards and PHA settings confirmed.
- ◆ Standard count rates acquired on primary and secondary standards. Evaluate accuracy with secondary standards.
- ◆ Locations on samples identified using backscattered-electron imaging and energy-dispersive spectrometer, points digitized.
- ◆ X-ray measurements acquired on samples.
- ◆ ZAF (i.e., $\Phi(\rho z)$) corrections made on samples using assigned standards.
- ◆ Analyses inspected: total, stoichiometry, etc.
- ◆ Run finished, data dumped to Excel.

What Materials Are Analyzed by EPMA?

Sample Requirements

- ◆ Samples must be:
Solid (EPMA does not include VP conditions)
Polished flat (~0.25 μ m)
Conductive or made so by evaporation of ~25 nm carbon
Stable in vacuum and under electron beam bombardment
- ◆ Materials historically analyzed:
Elements Be-U
Inorganics, metals/alloys, ceramics, minerals, glasses
Biological, pharmaceutical, organic, etc. if fixated/stabilized
- ◆ Bulk materials (homogeneous within scattering volume)
Also particles, thin films/multilayers, rough surfaces, tilted
All deviations from flat and normal to beam must be treated with specific modifications to correction algorithms

Sample Preparation Issues

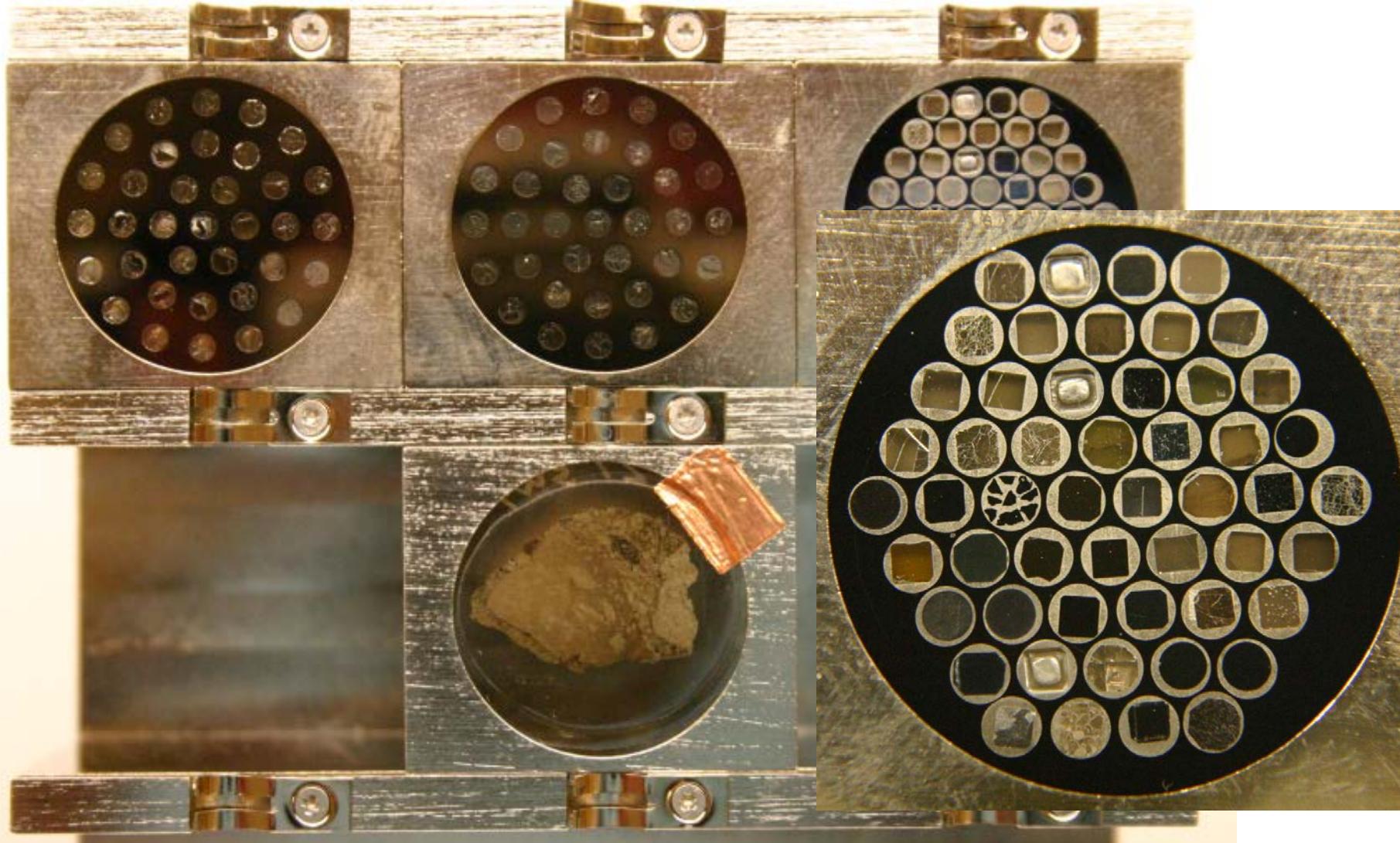
Polishing and Coating

- ◆ EPMA requires flat polished sample, usually mechanical polishing to ~0.25 um final polish.
- ◆ Biological materials other preparations methods, surface topography, results in differential x-ray absorption
- ◆ SEM imaging use Au or metal sputter coating or heavy metal fixation for high SEI emission.
 - Not preferred for microanalysis due to thickness variation
 - Vacuum ~ 10^{-3} Torr, thermal damage of biological matls.
- ◆ EPMA microanalysis use carbon high vacuum evaporation
 - Desire equal coat on samples and standards
 - Minimum x-ray absorption most x-rays (except Nitrogen)
 - Electron retardation affects overvoltage of energetic x-rays
 - Vacuum ~ 10^{-6} Torr, no thermal damage except radiant

High Vacuum Carbon Evaporator



Microprobe Sample Holder and Standard Mount



EPMA Summary
Matrix Effects
and
Quantitative EPMA

Summary of EPMA

- ◆ “Electron-probe microanalysis”, EPMA, generically means:
 - Quantitative X-ray microanalysis using a focused electron beam
 - Wavelength-dispersive spectrometers (WDS)
 - Optical microscope for sample positioning
 - Energy-dispersive spectrometer (EDS), ...*not standardless*
 - Stabilized electron column (gun shift, tilt, probe current stabilization)
- ◆ Characteristic x-rays measured on flat, polished (bulk) material at known x-ray takeoff angle to spectrometer. Elements Be-U at ~0.01-100 wt%
- ◆ Peak intensities background and deadtime corrected, precision
 $\sigma = \text{sqrt}(N)$ for N number of counts.
- ◆ Comparative technique: Ratio of x-ray intensity on sample is made to same on standard (“k-ratio”) using peak intensities with background subtracted:
$$k = (P-B)^{\text{sample}} / (P-B)^{\text{standard}}$$
- ◆ Measured x-ray intensities are converted to concentration units via:
$$C_{\text{sample}} = C_{\text{std}} K * Z A F, \quad K = (P-B)^{\text{sample}} / (P-B)^{\text{standard}}$$
The ZAF correction is iterative since it is a function of composition.
- ◆ Issues: sample volume, standards, background determination, secondary fluorescence, accuracy.

EPMA Summary: ZAF and X-ray Correction

- ◆ The absorption factor A corrects for x-ray absorption.

Beers Law absorption:

$$I = I_0 e^{(-\mu/\rho \rho_s)}$$

I emitted X-ray intensity, I_0 generated X-ray intensity

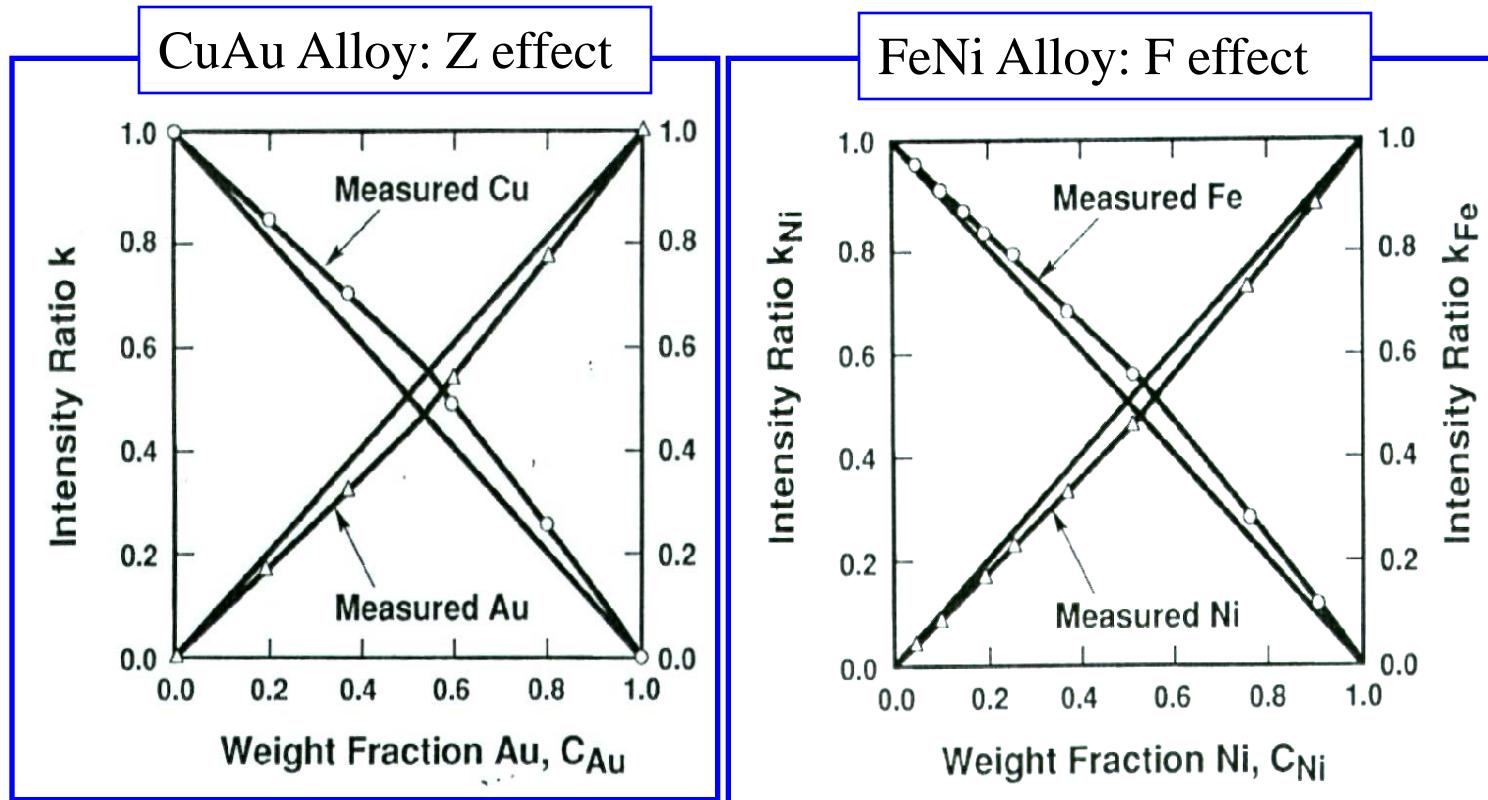
Path length ρ_s and mass absorption coefficient μ/ρ

- ◆ Absorption increases with MAC and path length (...TOA, keV)
MACs Al absorber: Cu L α 1452, Al K α 402, Cu K α 47
MACs Cu absorber: Cu L α 1590, Al K α 4561, Cu K α 50
- ◆ Absorption generally decreases with increasing X-ray energy, except at absorption edge of matrix element(s).
- ◆ Absorption occurs within all materials, sample and standard.
- ◆ This is the largest correction ~10% for typical materials.
 - AlCu alloy: Al K α absorbed by Cu
 - NiFe alloy: Ni K α absorbed by Fe
 - Fe K α fluoresced by Ni K α
- ◆ We measure **Emitted** intensity (I) but **Generated** (I_0) intensity is required for quantitative correction.
- ◆ Need to know generated intensity, unseen, unmeasured.
...Fundamental reliance on algorithms and data sets

EPMA Summary: ZAF and X-ray Correction

- “Matrix effects”: All elements influence electron scattering/retardation, x-ray generation, absorption, and fluorescence of the element of interest, i :

$$C_i = k_i * \text{ZAF} \text{ where ZAF is effect of } C_j \text{ elements on } C_i$$



EPMA Summary: ZAF and X-ray Correction

- ◆ Matrix effects arise because each element present in the analytical volume influences x-ray generation and propagation of every other element:
 - C_i is a function of all C_j elements
 - $C_i = k_i * \text{ZAF}$ where ZAF is effect of C_j elements on C_i
- ◆ Z: atomic number factor Z corrects for differences in
 - (S) stopping power, i.e., electron retardation
 - (R) electron backscattering...between the sample and standard.
 - Example: Al K α in Al₃Cu₉₇ with avg. Z = 28.8
 - vs. pure element standards: Al Z = 13, Cu Z = 29
 - Expect large Z correction when matls. differ in atomic number
- ◆ A: absorption factor A corrects for x-ray absorption within both the sample and standard. This is the largest correction parameter, and is critically dependent on accelerating voltage
 - Examples: Ni K α absorbed by Fe, Al K α abs by Cu

EPMA Summary: ZAF and X-ray Correction

- ◆ Z: atomic number factor Z corrects for differential (S) stopping power and (R) backscattering between the sample and standard.
Example: Al K α in Al₃Cu₉₇ with avg. Z = 28.8
vs. pure element standards: Al Z = 13, Cu Z = 29
Large Z correction when large difference in Z
- ◆ Note that R/S compensates so Z factor is not typically large
- ◆ ZAF factors typically presented as multiplicative factors

CalcZAF Results for Al₃Cu₉₇ Sample (also shows large A correction for Al K α) calculated at 15 keV and 40 degree takeoff angle:

	A	F	Z	ZAF	S	R			
ELEMENT	ABSCOR	FLUCOR	ZEDCOR	ZAFCOR	STP-POW	BKS-COR	F(x)u	Ec	Eo/Ec
Cu ka	1.0000	1.0000	1.0024	1.0024	1.0032	.9992	.9899	8.9790	1.6706
Al ka	2.5365	1.0000	.8904	2.2586	.7872	1.1311	.3498	1.5600	9.6154
ELEMENT									
	K-RAW	K-VALUE	ELEMWT%	OXIDWT%	ATOMIC%	FORMULA	KILOVOL		
Cu ka	.00000	.98471	98.704	-----	97.000	.970	15.00		
Al ka	.00000	.00574	1.296	-----	3.000	.030	15.00		
TOTAL:			100.000	-----	100.000	1.000			

$$C = k * \text{ZAF}, \text{ for Al: } C = 0.00574 * 2.2586 = 0.01296 * 100\% = 1.296 \text{ wt\%}$$

EPMA Summary: ZAF and X-ray Correction

- ◆ NiFe Alloy, example of (A) absorption and (F) characteristic fluorescence

Ni K α absorbed by Fe, note MAC value 362

Fe K α fluoresced by Ni K α , note F value 0.78

Current Mass Absorption Coefficients From Henke (LBL, 1985) < 10KeV / CITZMU > 10KeV

Z-LINE	X-RAY	Z-ABSOR	MAC
Ni	ka	Ni	57.8
Ni	ka	Fe	362.3
Fe	ka	Ni	89.3
Fe	ka	Fe	68.3

ELEMENT	ABSCOR	FLUCOR	ZEDCOR	ZAFCOR	STP-POW	BKS-COR	F(x)u	Ec	Eo/Ec
Ni ka	1.0060	1.0000	.9987	1.0047	.9994	.9993	.9824	8.3330	1.8001
Fe ka	1.0039	.7768	1.0135	.7903	1.0056	1.0079	.9808	7.1120	2.1091

ELEMENT	K-RAW	K-VALUE	ELEMWT%	OXIDWT%	ATOMIC%	FORMULA	KILOVOL
Ni ka	.00000	.90021	90.441	-----	90.000	.900	15.00
Fe ka	.00000	.12095	9.559	-----	10.000	.100	15.00
TOTAL:			100.000	-----	100.000	1.000	

EPMA Summary
Electron-Specimen Interactions
and
Generation and Emission of X-rays

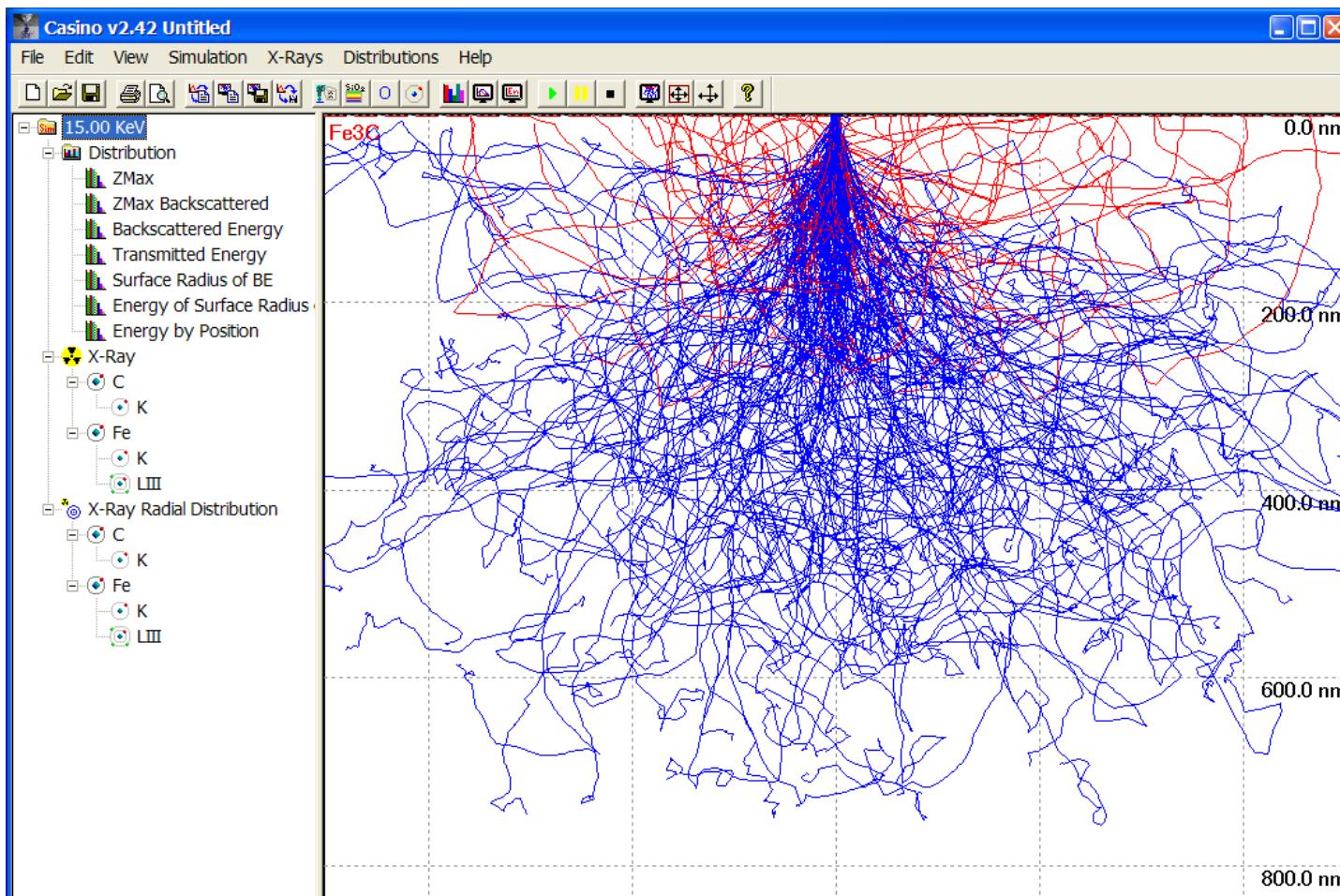
Resolution in EPMA

- ◆ There are two aspects to “resolution” in EPMA:
- ◆ Electron scattering and the emission of secondary and backscattered electrons govern the SEI and BSE image signals
- ◆ The scattering volume is a function of average atomic number Z_{avg}
- ◆ The x-ray generation “volume” is a contoured volume depending on the excitation energy E_c for a given x-ray line
- ◆ Compare Cu Ka and Al Ka in an AlCu alloy

Monte Carlo Modeling

- ◆ Monte Carlo simulations model electron-specimen interactions and x-ray production in the specimen.
- ◆ Electron scattering, slowing down, backscattering, x-ray production, etc. are all modeled using random number generation and random electron scattering directions.
- ◆ Good agreement is observed between Monte Carlo data and experimental measurements.
- ◆ This agreement is evidence of the accuracy of quantitative analysis algorithms.

Casino Monte Carlo Program



Simulation Fe₃C, 15 keV, 5000 trajectories, max depth ~750nm

Red = backscattered e-, blue e- within Fe₃C.

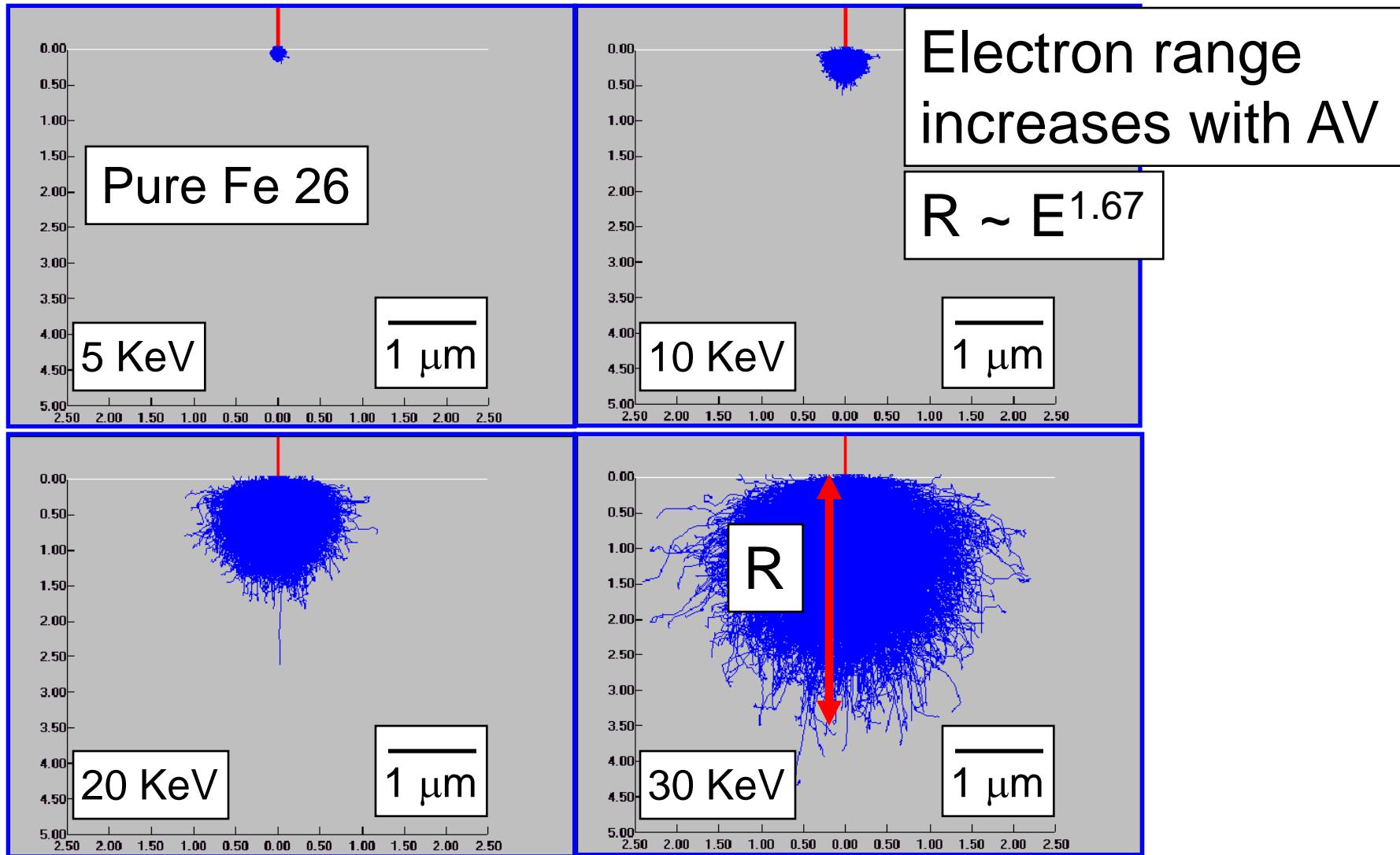
Scattering angle increases as e- energy decreases (curly trajectory)

Monte Carlo Software - Casino, DTSA II

- ◆ Casino:
 - Scattering volume
 - $\Phi(\rho z)$ curves for generated vs. emitted X-rays (K vs. L-lines)
 - Electron data, e.g. BSE energy distribution for Al vs. Au targets
 - Bulk vs. Layer geometry or grain boundary
- ◆ DTSA-II
 - EDS spectrum simulation: Experimental vs. Calc Spectra
 - Monte Carlo vs. Algorithmic X-ray simulation
 - Powerful scripting capabilities: Comp, Geometry, Structure

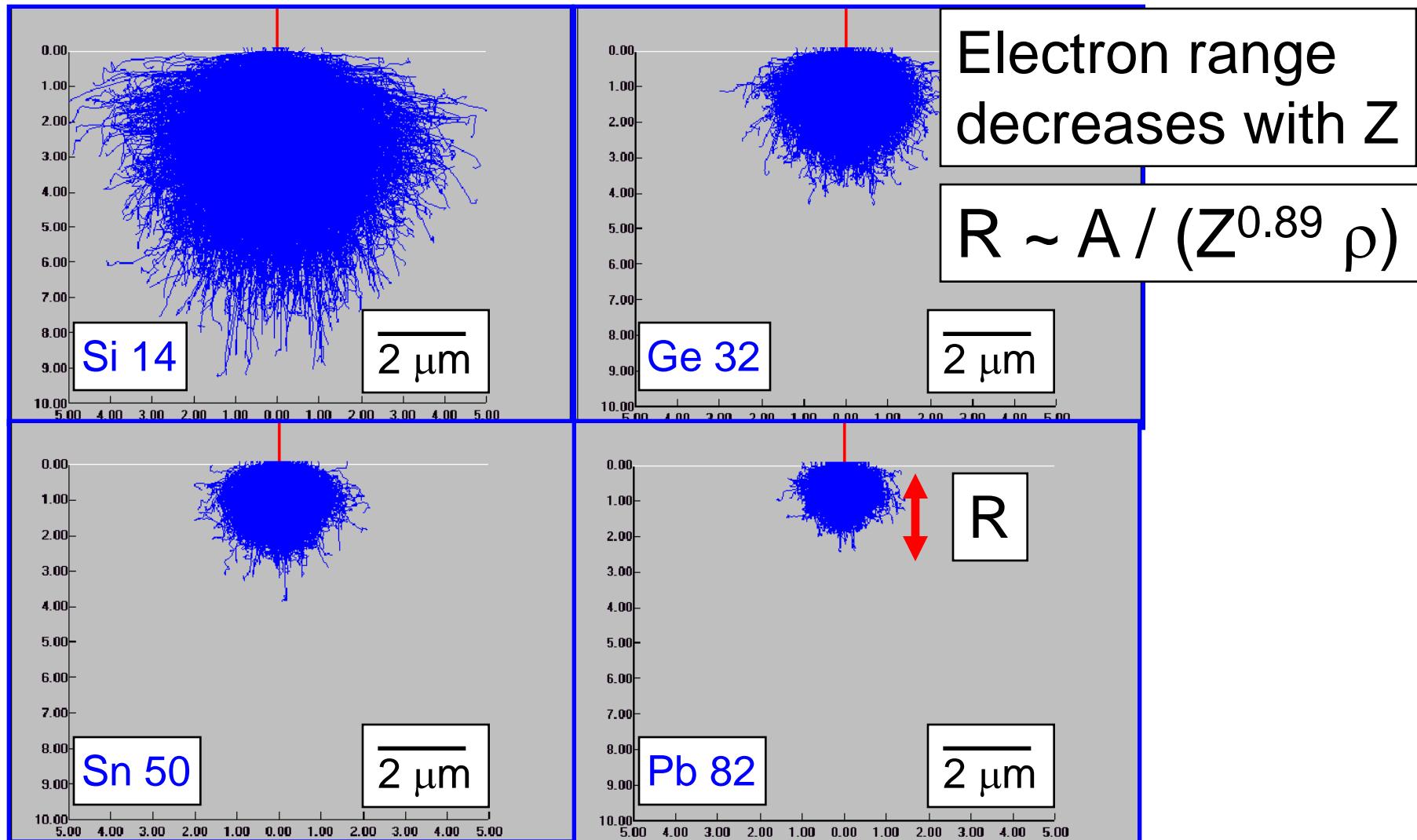
Monte Carlo Simulation

Effect of Varying Accelerating Voltage

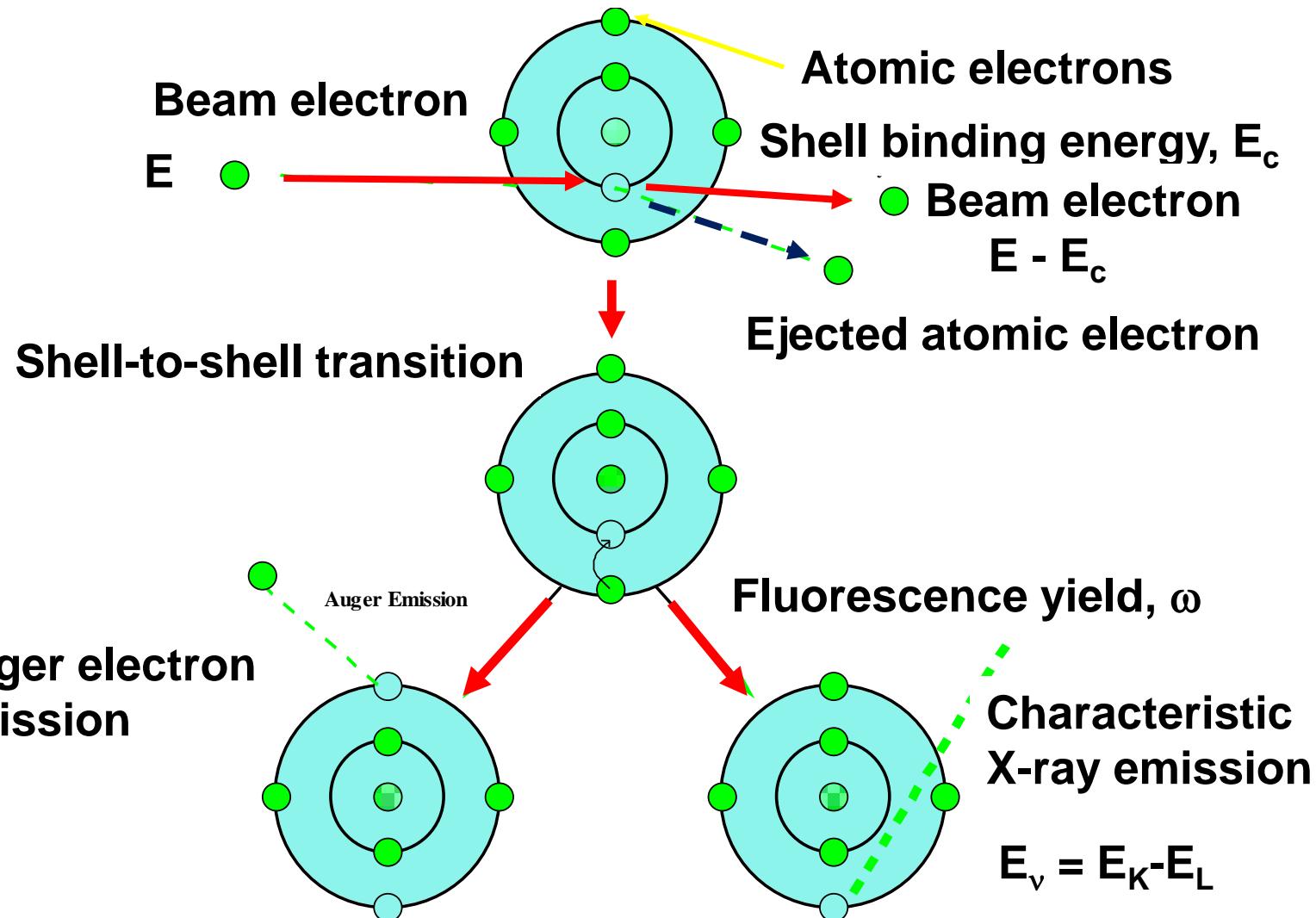


Monte Carlo Simulation

Effect of Varying Atomic Number Z @ 25 keV



The Physics Underlying the Spectrometries



Families of X-rays

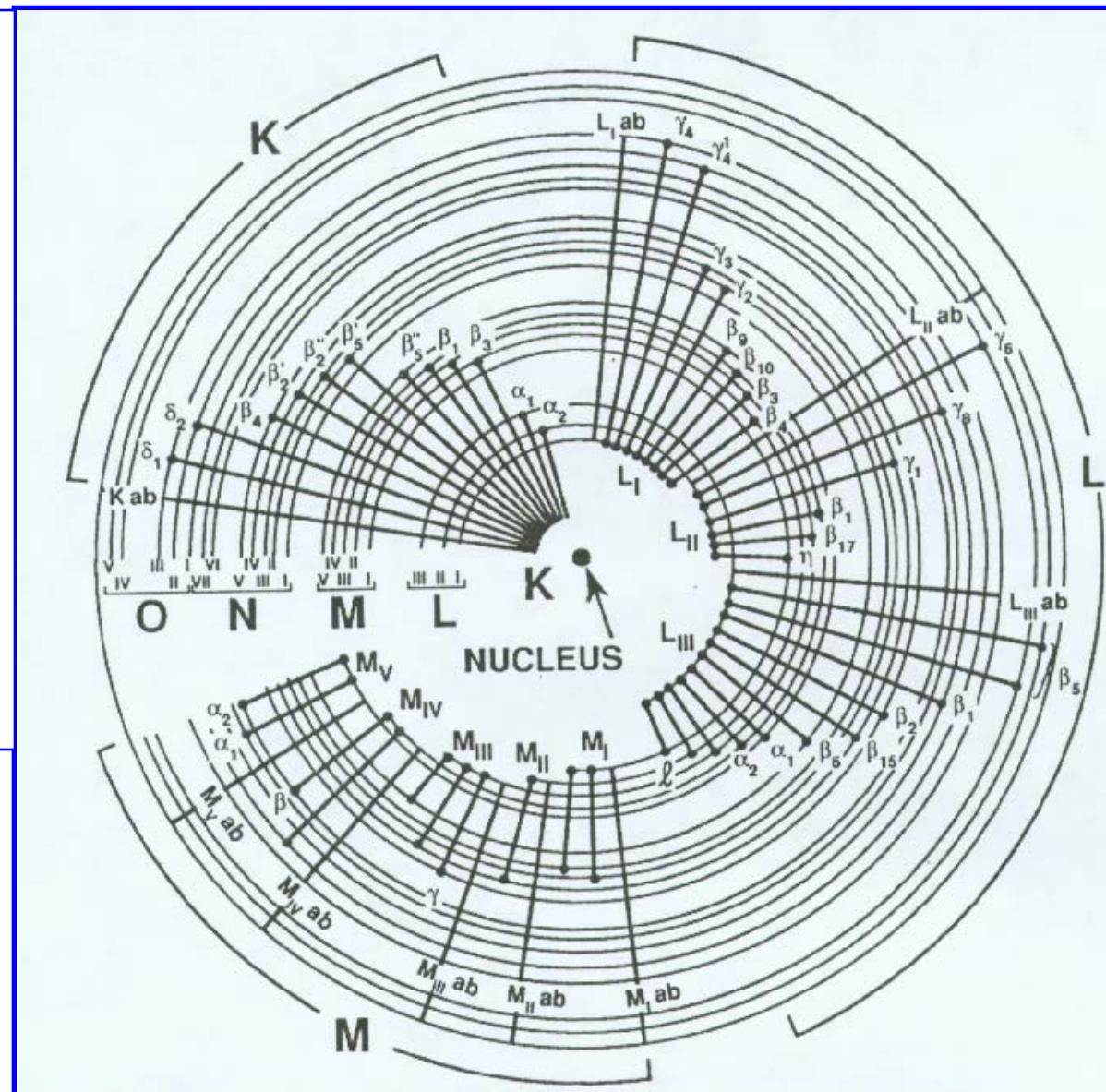
X-ray Spectra

Many ionization
vacancy-filling
possibilities from
complex atoms lead to
creation of families of
X-ray lines, e.g.,

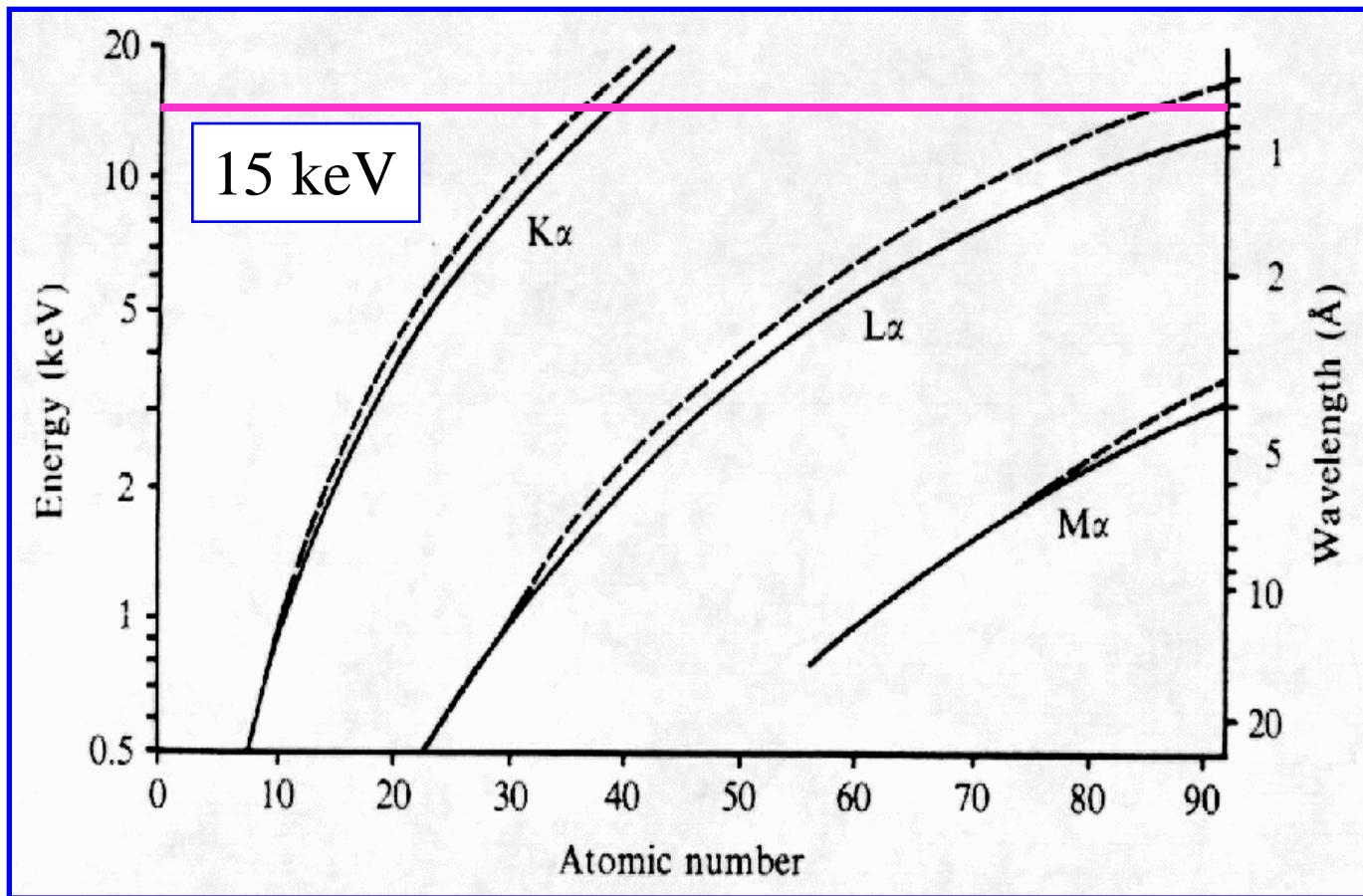
$K\alpha$ - $K\beta$

$L\alpha$ - $L\beta$ - $L\gamma$ - $L\eta$ - $L\iota$

$M\alpha$ - $M\beta$ - $M\gamma$ - $M\zeta$ - M_{II} N_{IV}

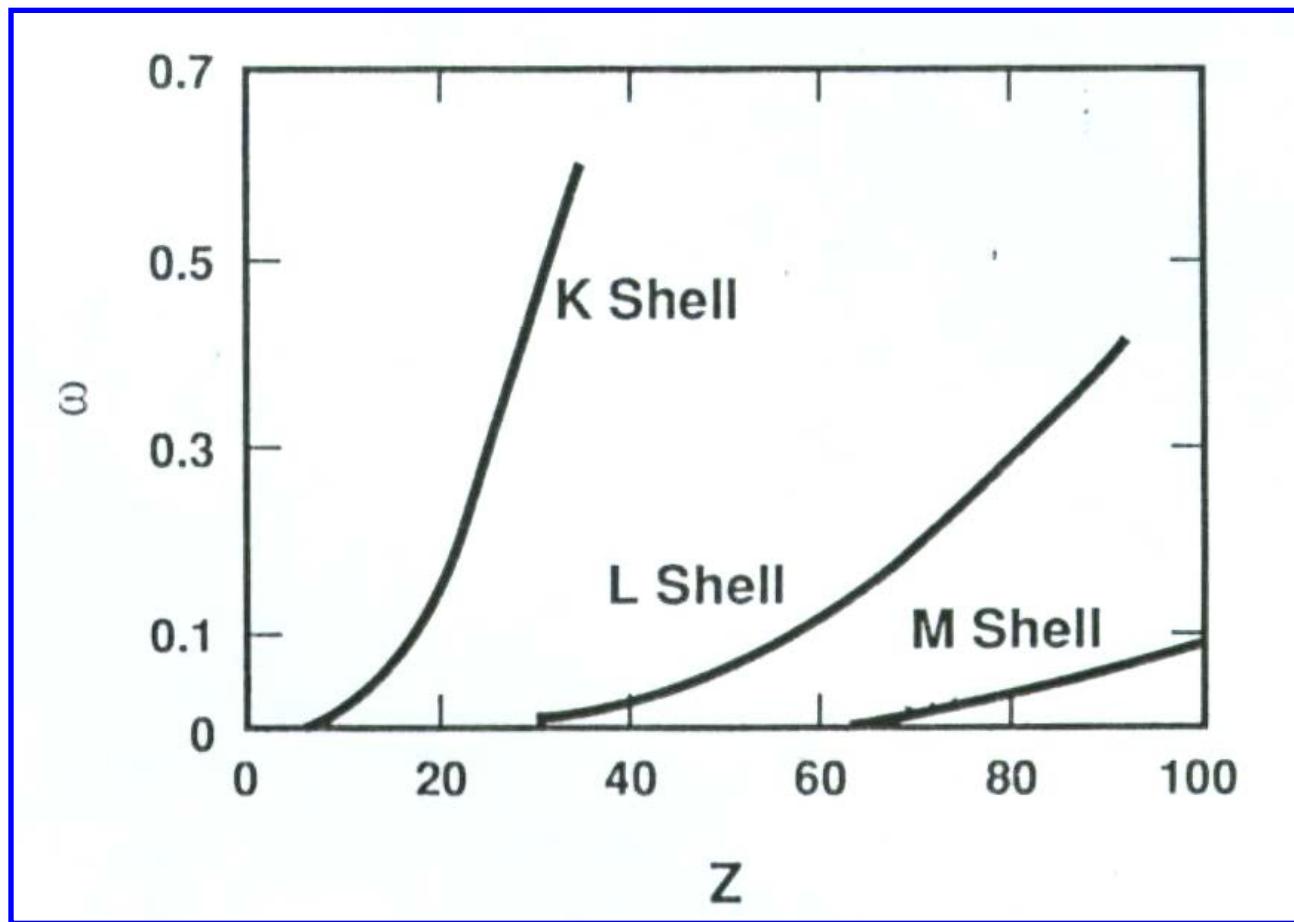


Characteristic X-ray Energy vs. Z



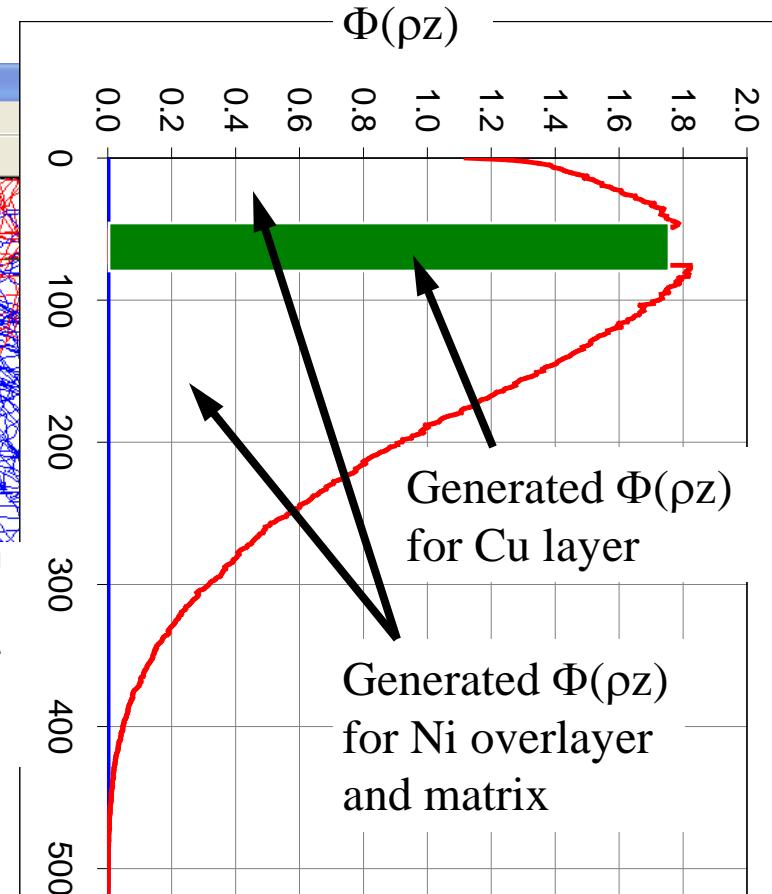
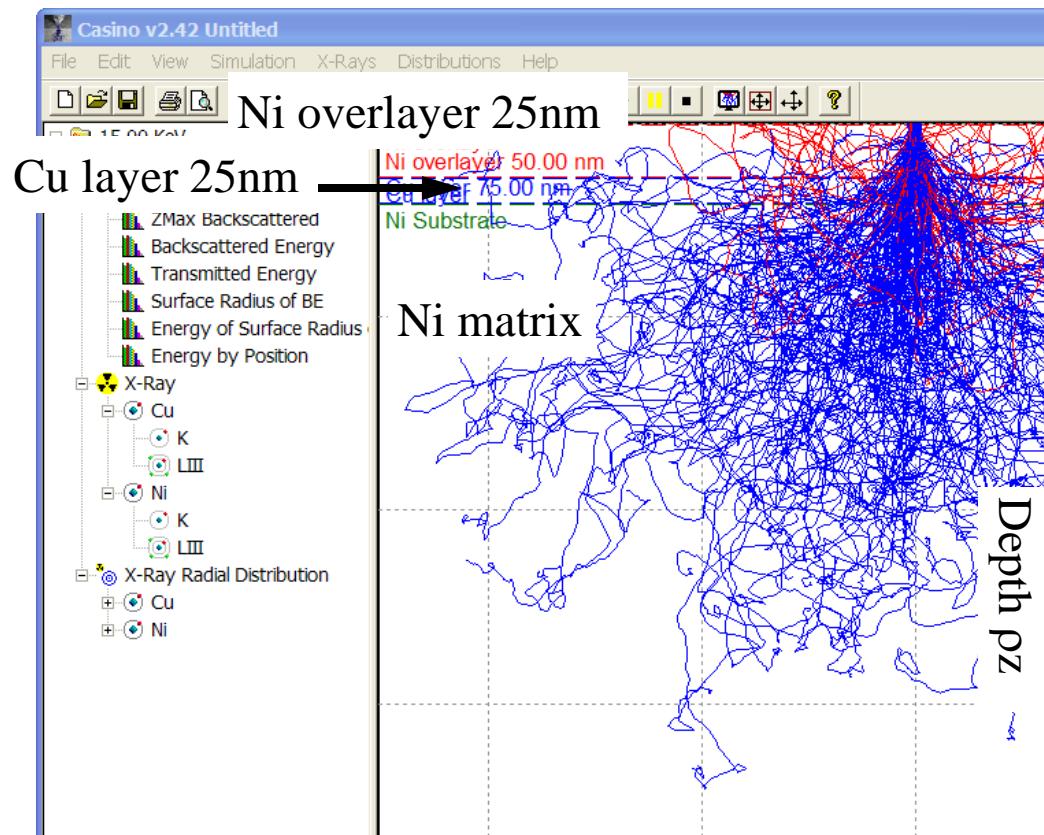
X-ray peak energy solid lines
Excitation energy dashed lines
At 15 keV at least one line can be excited for all Z

Fluorescence Yield ω vs. Z



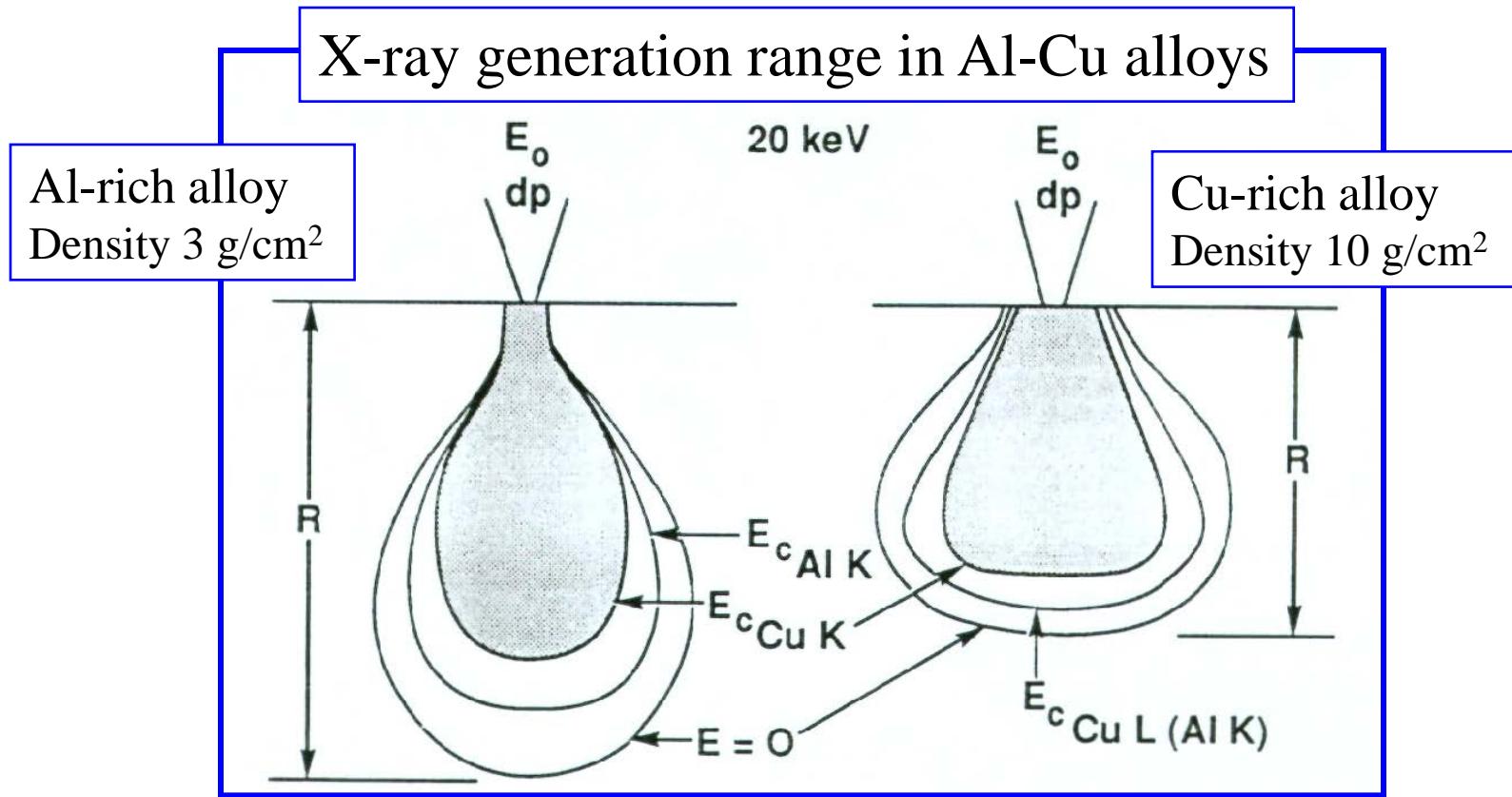
ω = fraction of ionizations resulting in characteristic x-ray emission relative to total number of ionizations, and $1-\omega$ is the fraction of auger electrons.

Casino MC Simulation of $\Phi(\rho z)$ Experiment



The $\Phi(\rho z)$ distribution for Cu $K\alpha$ is obtained from the emitted intensity of a layer of Cu sandwiched in a matrix with similar Z (Ni) so that scattering properties are similar. Measurements are made for progressively buried layers. This intensity is scaled relative to that from an isolated Cu foil to yield the $\Phi(\rho z)$ curve. The curves are then generalized to model x-ray behavior for elements in general.

Range of Characteristic X-ray Production



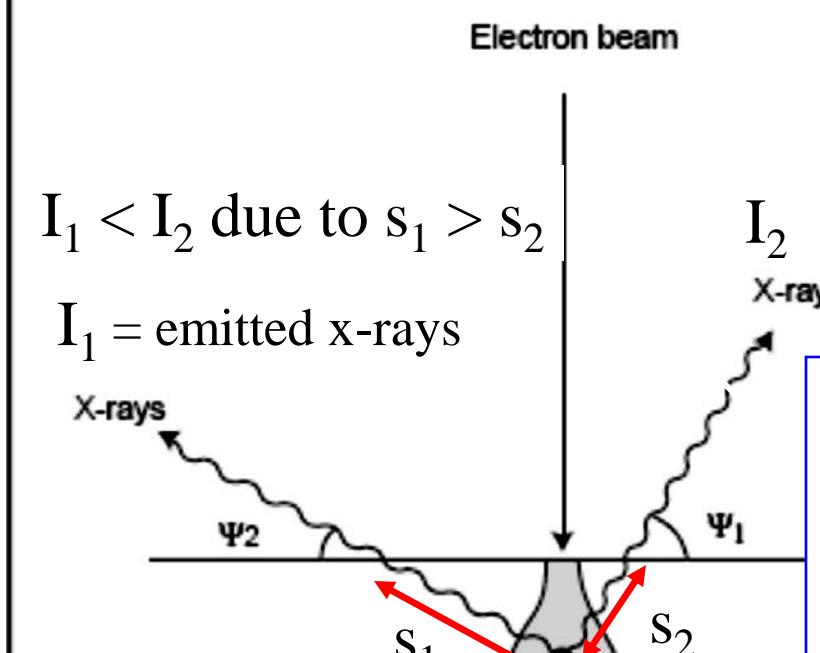
The depth distribution of primary x-ray generation depends on Z_{avg} and the relative excitation energies E_c of the elements in the sample:

Al K α $E_c = 1.559$ keV generated in larger volume $E > 1.559$ keV

Cu K α $E_c = 8.98$ keV, generated in small volume $E > 8.98$ keV

X-ray absorption depends on path length and μ/ρ

$$I / I_0 = \exp[-(\mu/\rho) ps]$$

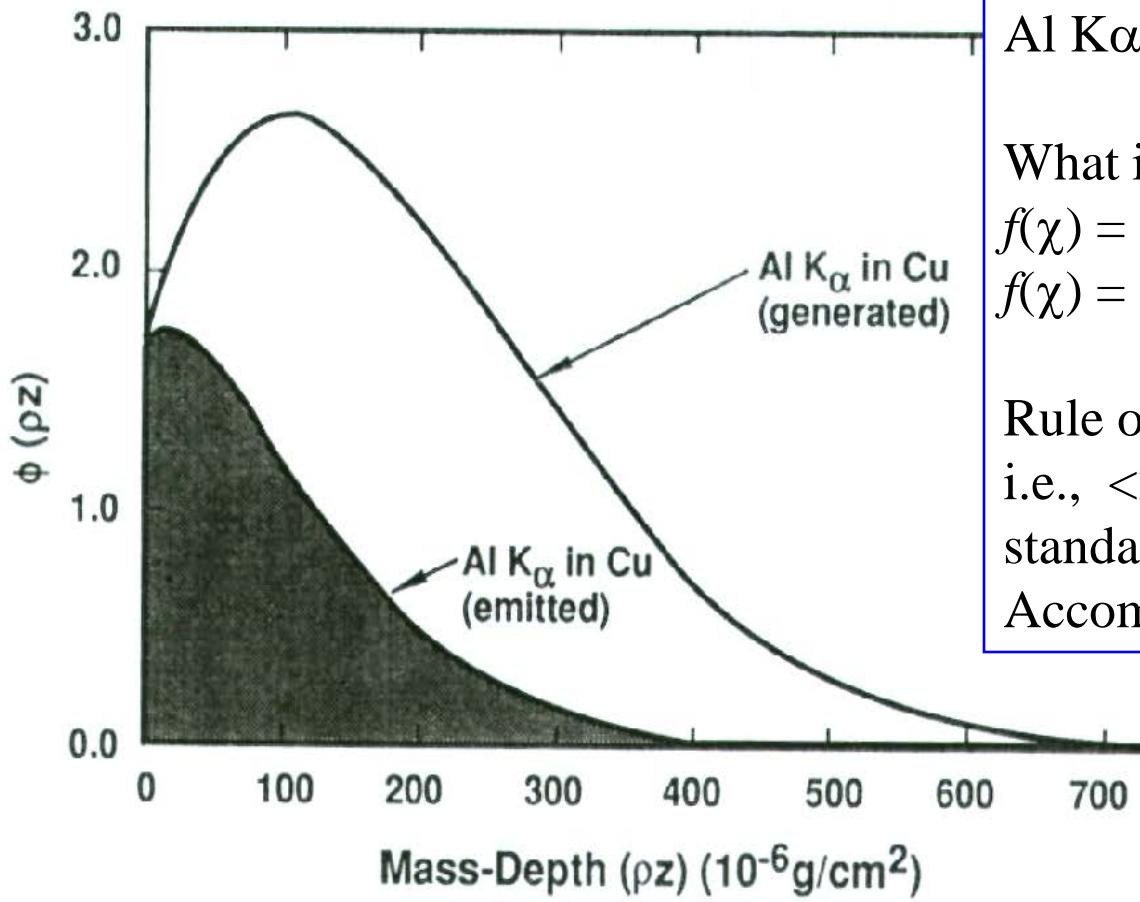


I_0 = generated x-rays

X-ray absorption here is photoelectric, which is an either/or operation. Either the photon is absorbed, or it escapes the specimen with the same characteristic energy with which it started.

For a given takeoff angle, the x-ray absorption is determined by the path length s and mass absorption coefficient μ/ρ , which are both compositionally dependent.

Effect of X-ray Absorption: Trace Al in Cu @ 20 keV



Al is strongly absorbed by Cu (μ/ρ Al K α by Cu = 4764).

What is $f(\chi)$?

$f(\chi) = \text{emitted} / \text{generated intensity}$

$$f(\chi) = I_{\text{emitted}} / I_{\text{generated}} = \sim 0.5$$

Rule of EPMA: $f(\chi) > 0.75$

i.e., <25% absorption in standard or sample

Accomplished by reducing kV

EPMA Summary: ZAF and X-ray Correction

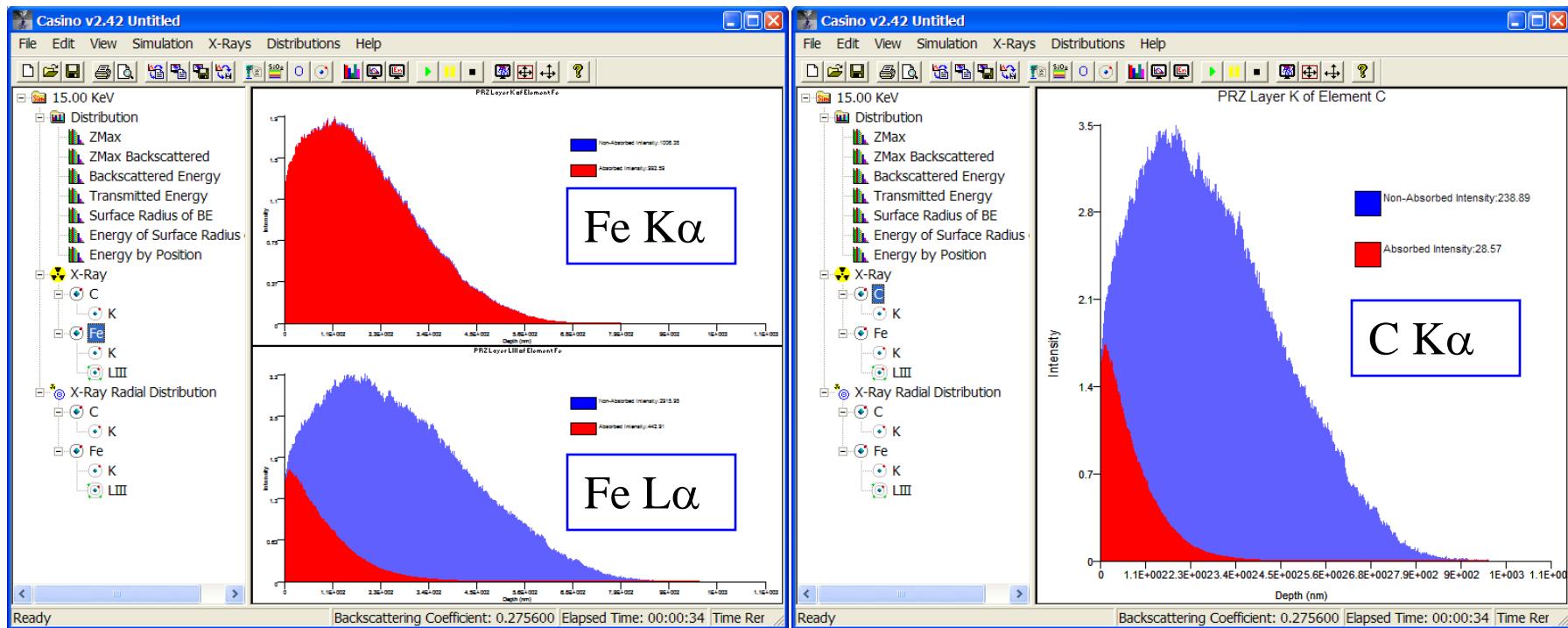
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Example: Al K α in Al₃Cu₉₇ with avg. Z = 28.8
vs. pure element standards: Al Z = 13, Cu Z = 29
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$$C = k * \text{ZAF}, \text{ for Al: } C = 0.00574 * 2.2586 = 0.01296 * 100\% = 1.296 \text{ wt\%}$$

Casino $\Phi(\rho z)$ Curves Calculated for Fe_3C



Casino Monte Carlo calculated $\Phi(\rho z)$ curves

Generated x-ray distribution in blue, Emitted x-ray distribution in red

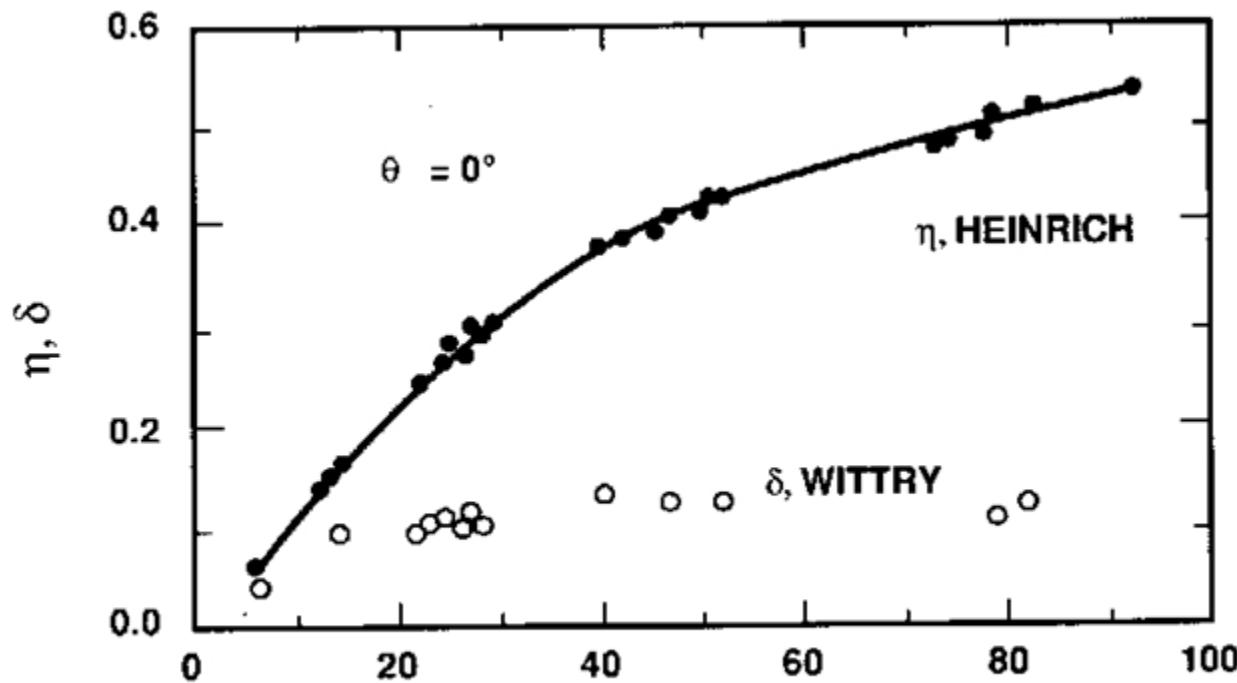
These intensities can be used to calculate k-ratios and compared with k-ratios from experimental and $\Phi(\rho z)$ algorithms

Backscattered Electron Imaging and X-ray Mapping

Contrast Mechanism

- ◆ Changes in the signal(s) (“Contrast”) collected at different points in the image convey information about specimen characteristics.
- ◆ Whenever a signal changes in a predictable way as a function of specimen properties, we have the basis for a contrast mechanism.
- ◆ Example: η vs. Z is monotonic and predictable. This forms the basis for atomic number (compositional) contrast.

Backscattered Electron Dependence on Z Atomic Number



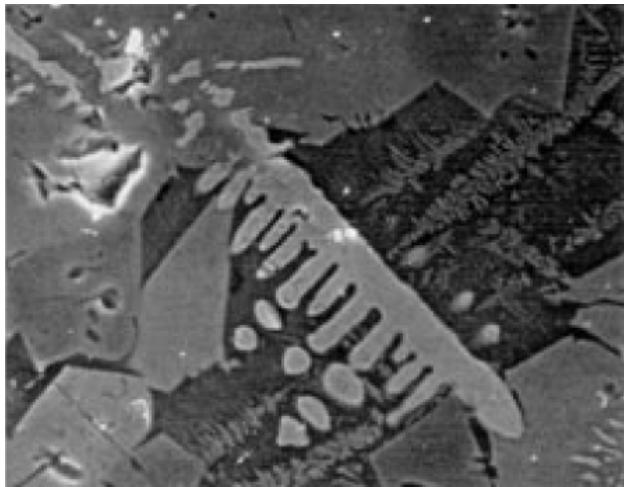
Backscattered electrons: primary beam electrons scattered at high angles by the nucleus of atoms in the sample. The fraction of BSE is eta and is a function of atomic number Z for pure elements and average Z for compounds. The BSE signal is used for mineral ID.

Secondary electrons are emitted from the valence shell of atoms and are not sensitive to the Z of the material, so are not useful for identifying phases

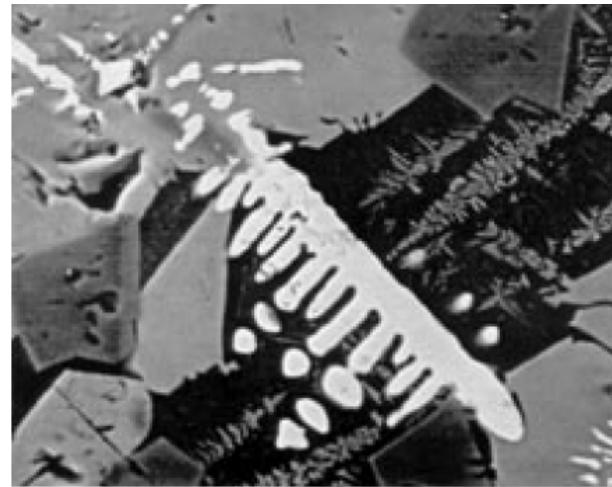
Mean Atomic Number of Minerals

- ◆ Mean Z can be easily calculated (CalcZAF, for example)
- ◆ Typical range is ~ 10 – 20. Atomic number correction is limited if appropriate standards are used.
- ◆ Silicates
 - Forsterite 10.6 Albite 10.7 Pyrope 10.7
 - Fayalite 18.7 Anorthite 11.9 Almandine 15.6
- ◆ Oxides
 - Periclase 10.4 Spinel 10.6 Quartz 10.8
 - Ilmenite 19.0 Ulvöspinel 20.0 Magnetite 21.0
- ◆ Sulfides
 - Pyrite 20.7 Chalcocite 26.4 Galena 73.2
- ◆ Carbonates
 - Dolomite 8.9 Calcite 12.4 Siderite 16.5 Cerussite 65.3
- ◆ Misc
 - Apatite 14.1 Allanite 22.0 Zircon 24.8 Monazite 38.7
- ◆ See SJB Reed Electron Microprobe Analysis and Scanning Electron Microscopy in Geology ISBN 0521483506 (paperback)

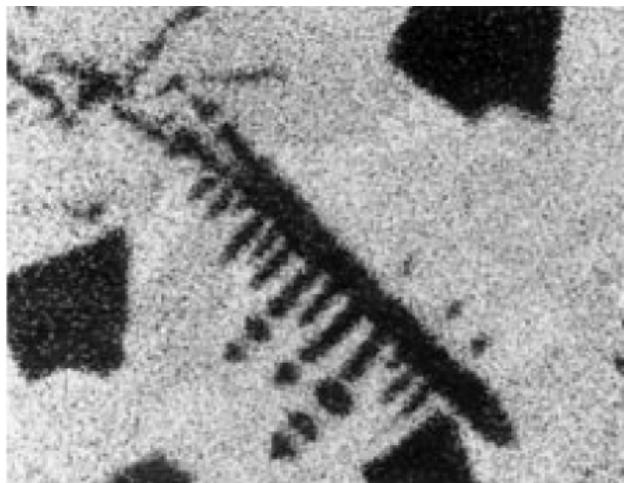
Image Signals from Metallographic Sample



Secondary electron



Backscattered electron



Silicon x-ray

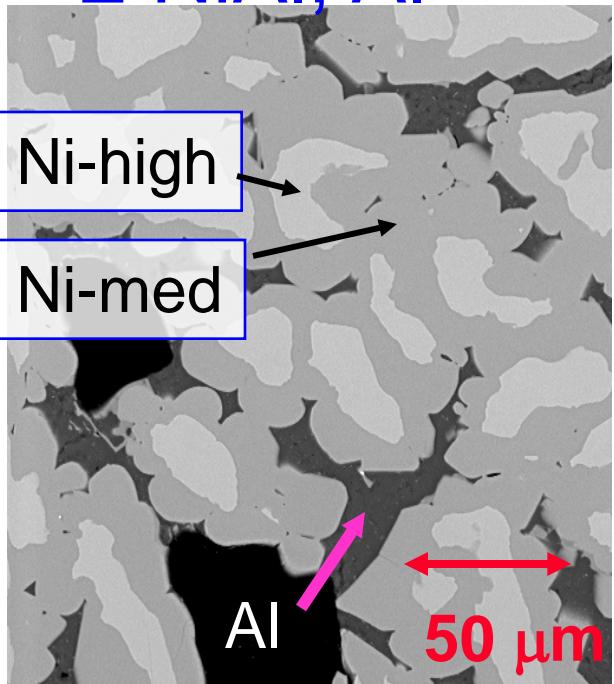


Aluminum x-ray

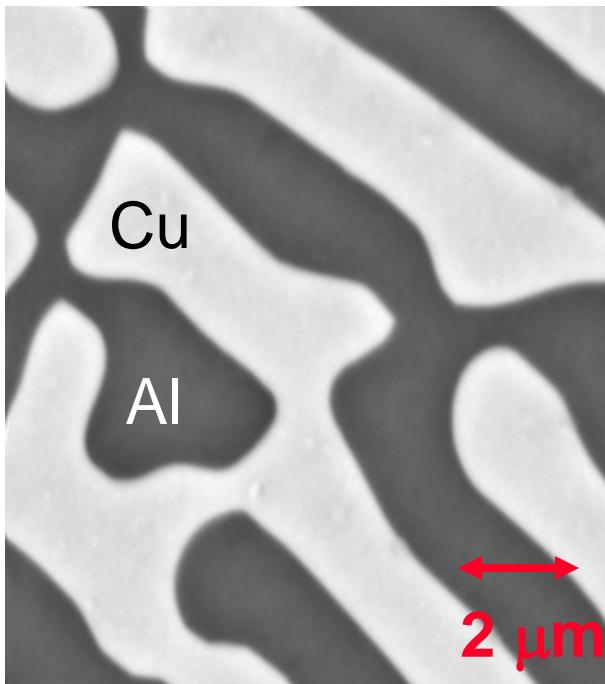
20 kV 1100x, JEOL

Atomic Number Contrast

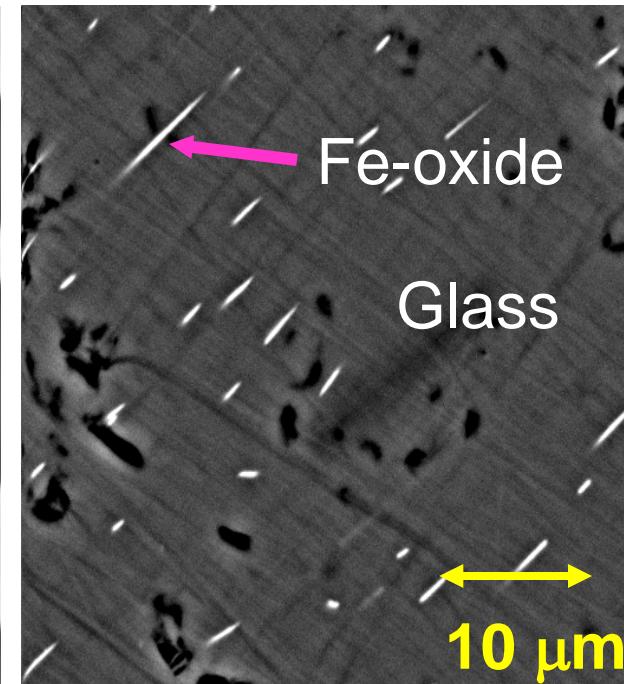
Raney Ni-Al
2 NiAl, Al



Al-Cu eutectic



Obsidian

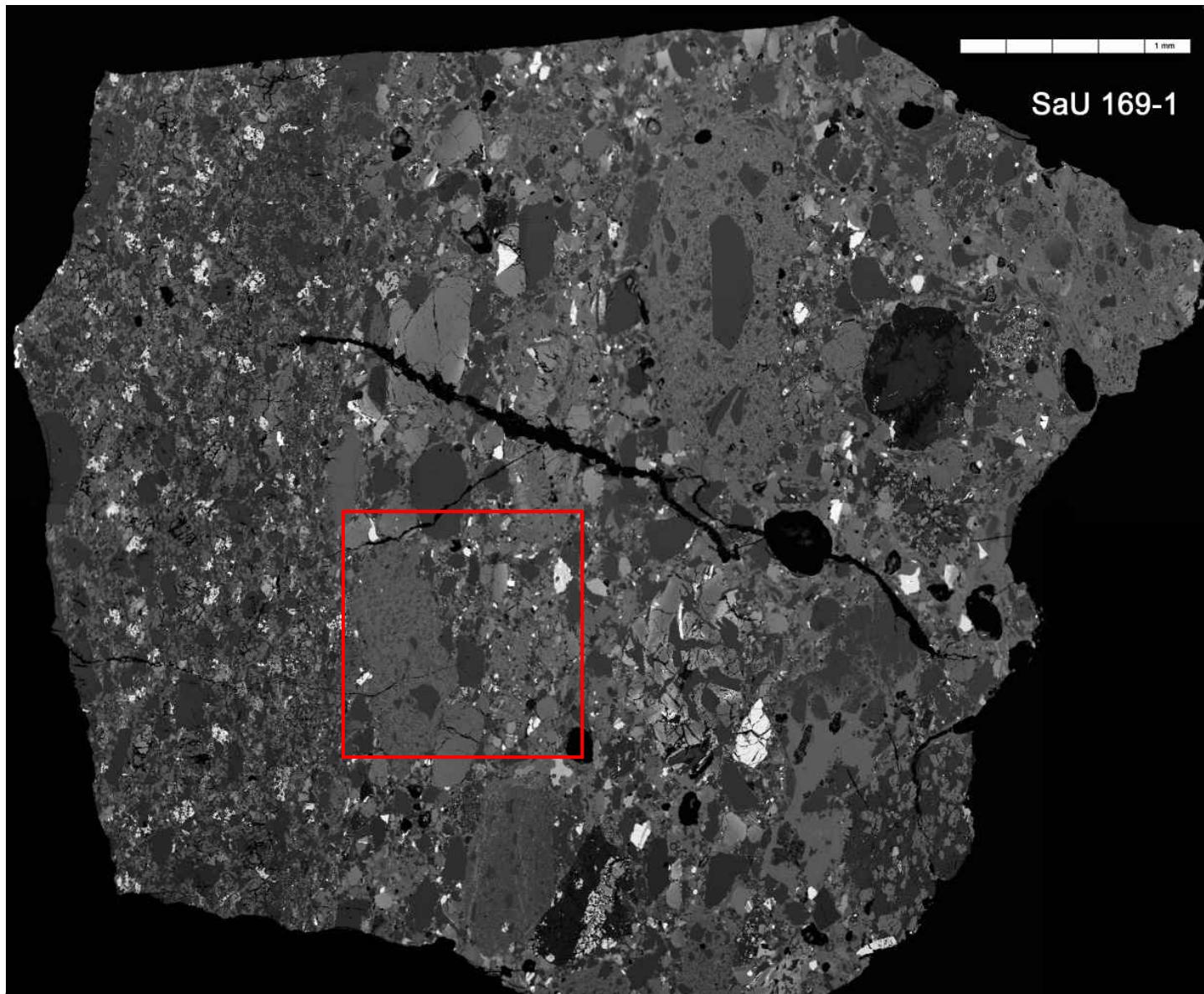


Backscattered-electrons provide a rapid and simple means to image differences in chemistry in the sample, via the dependence on average atomic number.

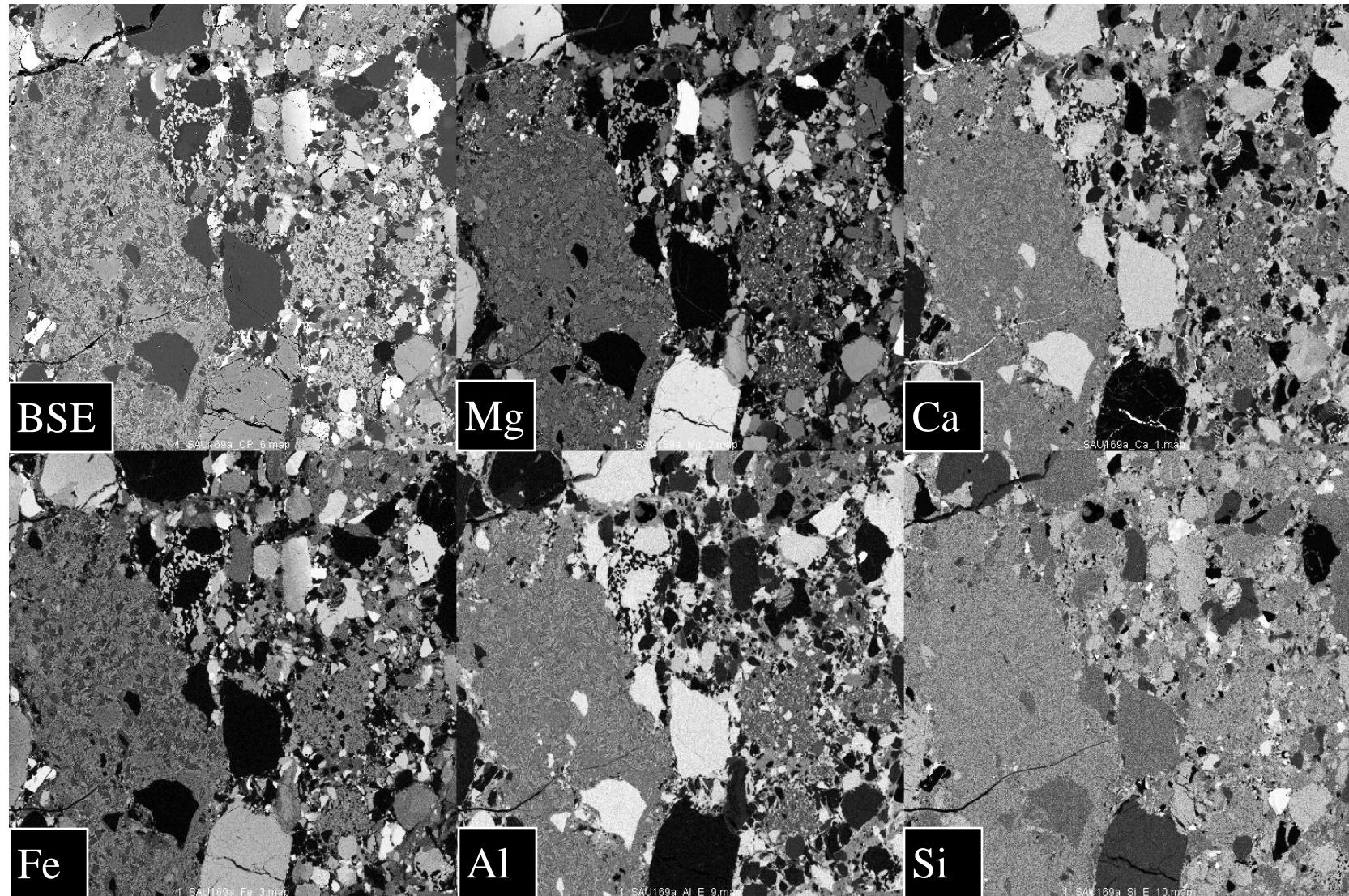
Compositional Mapping

- ◆ Normal x-ray mapping is performed by setting up an energy region of interest (ROI) in the EDS spectrum. All x-rays falling in that ROI are assigned to the digital map for that element.
- ◆ Drawback is each map has only information for that element and can not be used to perform quantitative analysis. X-rays from other elements also may be counted in the ROI – peak overlaps, continuum x-rays, etc.
- ◆ Sophisticated mathematical approaches have been used in recent years to process x-ray map data. These involve correlation between elements, but also correlation between groups of elements.
- ◆ Examples of these approaches are Concentration Histogram Imaging and Principle Component Analysis.
- ◆ Spectrum Imaging is a mapping technique where the entire EDS spectrum is saved at each pixel location and is available for processing after the run.

Lunar Meteorite SaU 169 BSE Mosaic



Lunar Meteorite SaU 169 SDD X-ray Maps

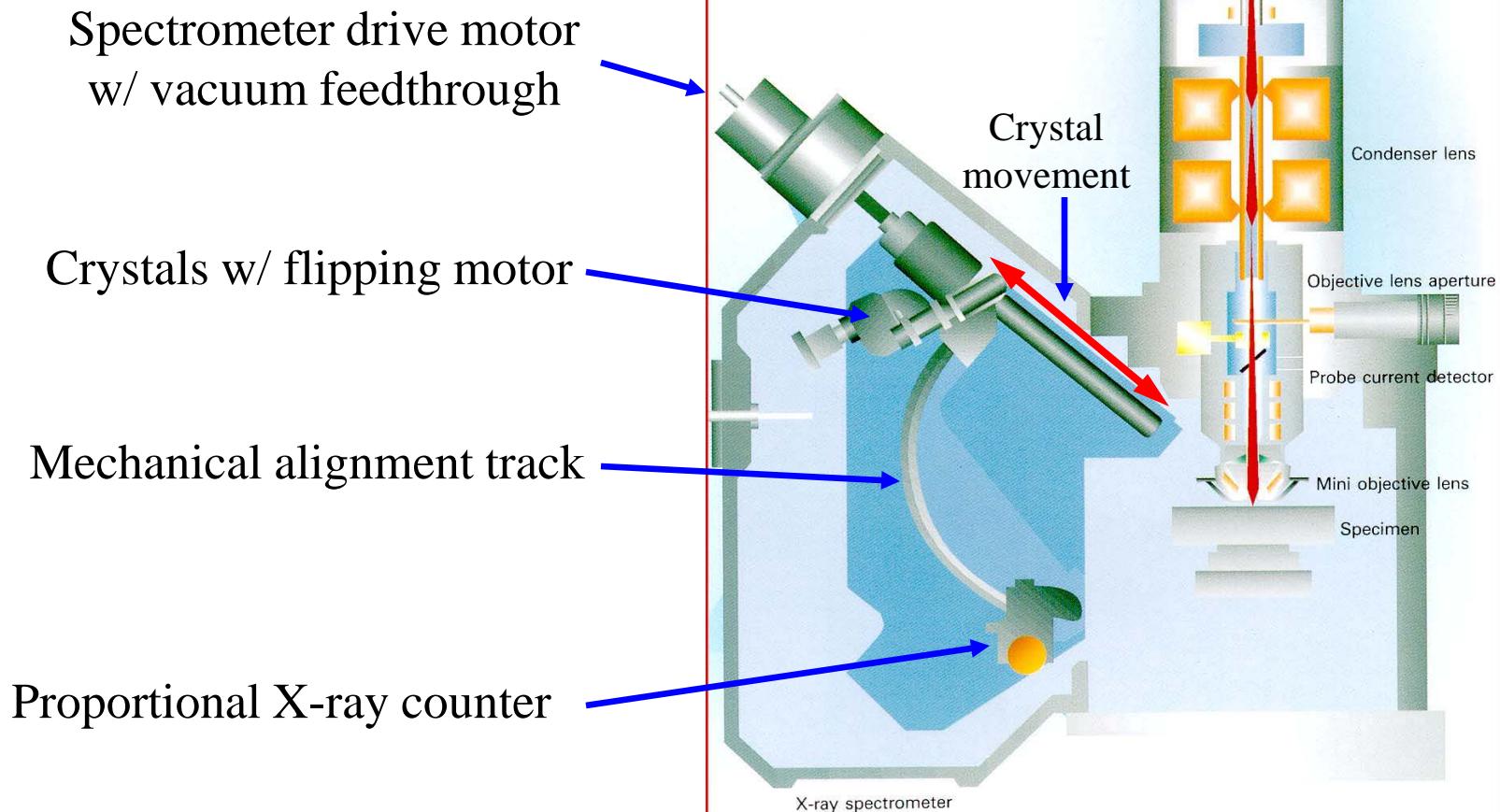


EPMA Summary

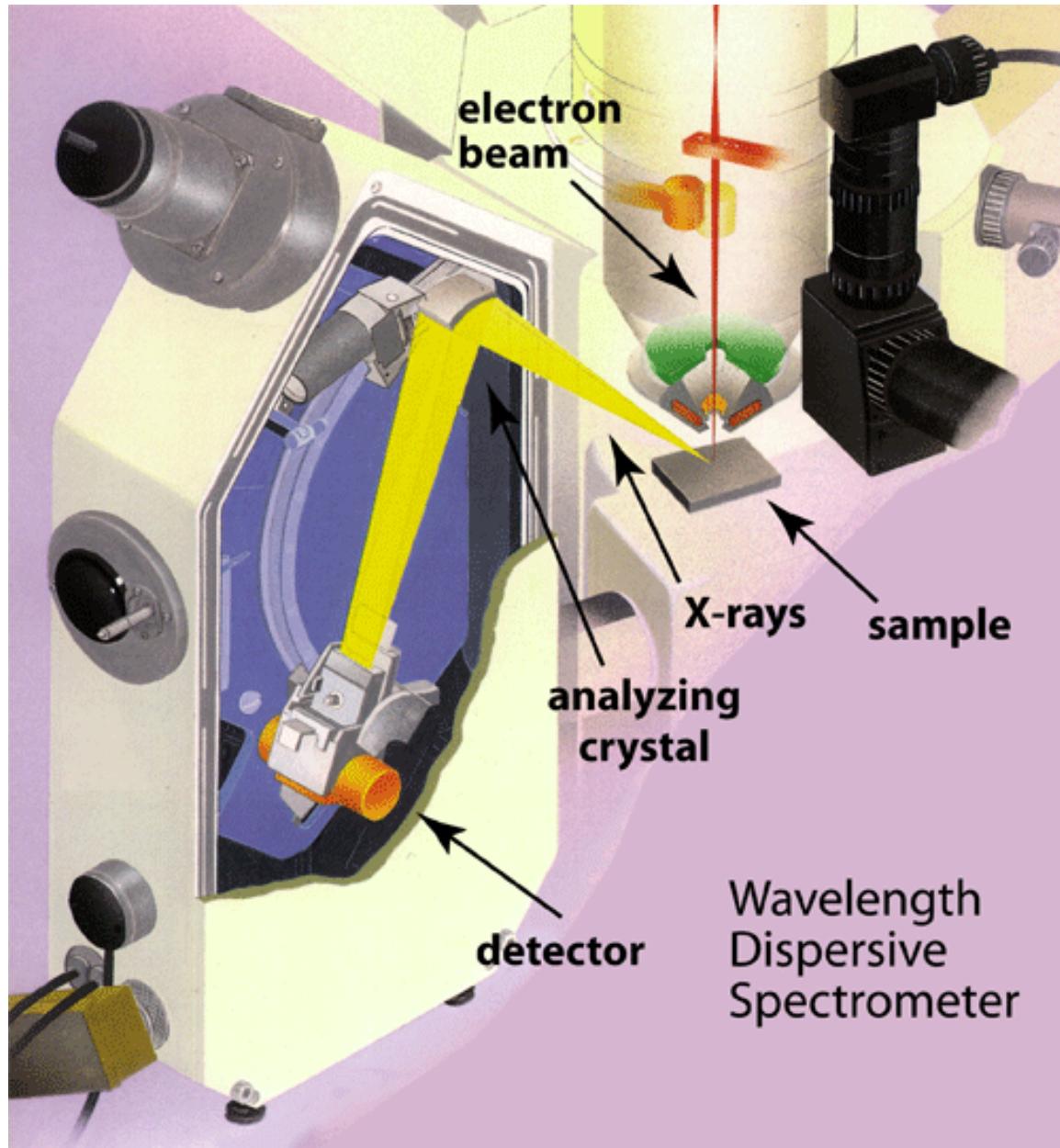
Wavelength-Dispersive Spectrometry

WDS Anatomy

An electron microprobe typically has up to 5 spectrometers each with 1-4 diffracting crystals. The crystal and detector are moved synchronously via the drive motor. Position here is high theta, low E.



WDS Schematic JEOL Electron Microprobe



WDS Spectrometer Units

- ◆ Bragg diffraction equation
 - including index of refraction term:
 $n\lambda = 2d \sin\theta (1 - k/n^2)$
 - n = diffraction order, λ = wavelength
 - d = d-spacing of analyzing crystal
 - θ = diffraction angle, k = refraction factor
- ◆ Cameca: $\sin\theta = n\lambda (1 - k / n^2) / 2d$
- ◆ For JEOL instruments:
 - $n\lambda/2d = \sin\theta = L/2R$ where R is rowland radius
 - $L = (R/d) n\lambda$, L value in mm

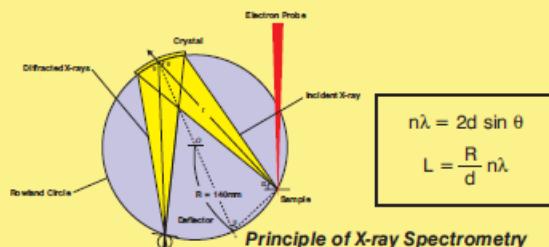
$$\text{JEOL: } L = (R / d) n\lambda (1 - k / n^2)$$

$$E = \frac{12.3985}{\lambda(\text{Angstroms})} (\text{keV})$$

L-Value X-ray Peak Positions for JEOL Microprobe

L-Value Table

	3Li	4Be	
LDEB K _α		221.00	
LDE2 K _α			
LDE1 K _α			
TAP K _α			
	11Na	12Mg	
TAP K _α	129.47	107.51	



5B	6C	7N	8O	9F	10Ne	
126.00 188.53	124.66 208.60	88.13 147.50	65.87 110.20 256.77	85.50 199.15	68.20 158.82	
13Al	14Si	15P	16S	17Cl	18Ar	
90.66	77.46 228.22	66.93 197.20	172.07	151.43	134.26	
n	31Ga	32Ge	33As	34Se	35Br	36Kr
9 11	93.18 122.75	87.20 113.45	81.77 105.13	76.82 97.73	72.30 91.04	68.15 84.98
d	49In	50Sn	51Sb	52Te	53I	54Xe
2	120.81	250.32 115.30	239.16 110.16	228.72 105.35	218.94 100.85	209.76 96.62
g	81Tl	82Pb	83Bi	84Po	85At	86Rn
1 9 9	83.96	81.71	79.54	77.45	75.45	73.52
1 9 9	174.88	169.31	163.94			
b	66Dy	67Ho	68Er	69Tm	70Yb	71Lu
4 1 1	132.73 61.14 104.25	128.29 100.01	124.07 95.88	120.07 92.18	116.26 88.59	112.61 85.23 251.11

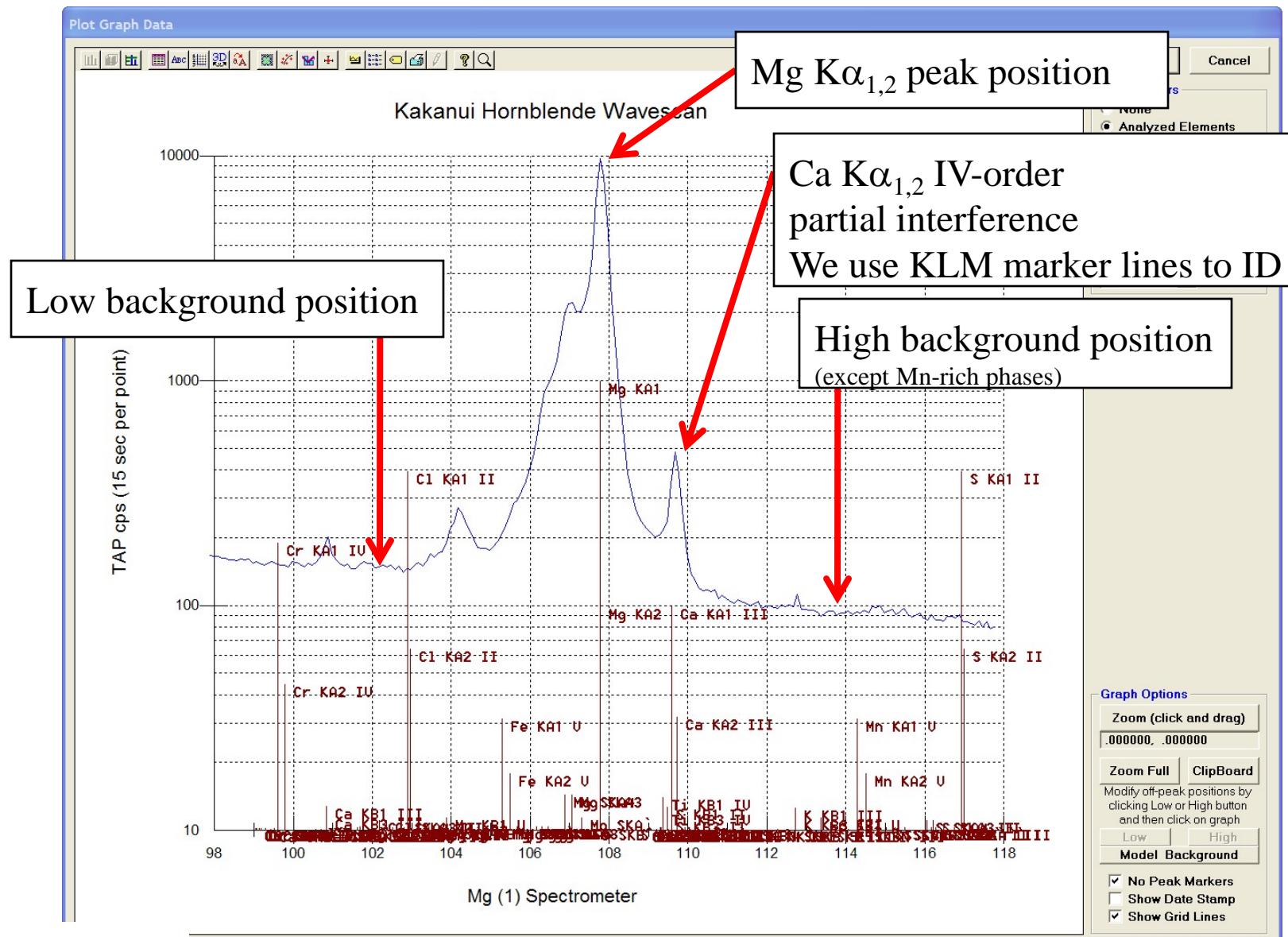
JEOL USA, Inc., 11 Dearborn Road, Peabody, MA 01960
Tel: 508-535-5900 Fax: 508-536-2205
Email: eod@jeol.com WWW: <http://www.jeol.com>

JEOL

Diffracting Crystal Element Ranges

	2d (nm)	6 C	14 Si	22 Ti	30 Zn	38 Sr	46 Pd	54 Xe	62 Sm	70 Yb	78 Pt	86 Rn	
TAP	2.576	8O	15P	24Cr	30Zn	38Sr	41Nb	46Pd	54Xe	62Sm	70Yb	78Pt	86Rn
TAPH	2.576	9F	13Al	24Cr	30Zn	35Br	41Nb	46Pd	47Ag	54Xe	62Sm	70Yb	79Au
PET	0.8742	13Al	18K	22Ti	25Mn	30Zn	36Kr	37Rb	41Nb	46Pd	54Xe	65Tb	70Yb
PETH	0.8742	14Si	18K	22Ti	30Zn	35Br	37Rb	41Nb	46Pd	56Ba	58Ce	72Hf	79Au
LIF	0.4027	19K	20Ca	31Ga	37Rb	48Cd	50Sn	54Xe	62Sm	70Yb	78Pt	86Rn	90Tl
LIFH	0.4027	20Ca	31Ga	37Rb	48Cd	50Sn	54Xe	62Sm	70Yb	78Pt	79Au	86Rn	90Tl

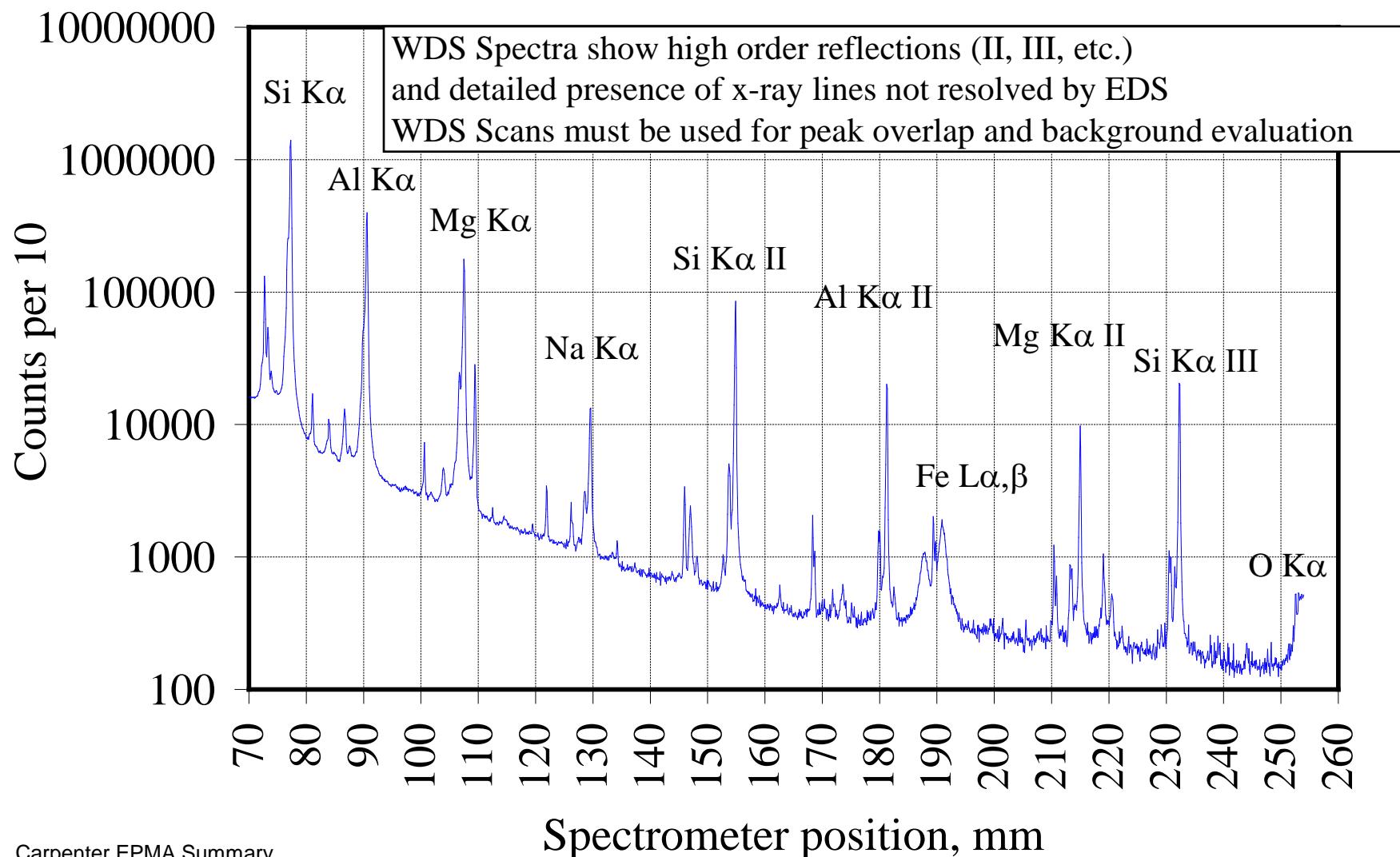
Mg K α peak using TAP crystal on Kakanui Hornblende



WDS Wavelength Scan

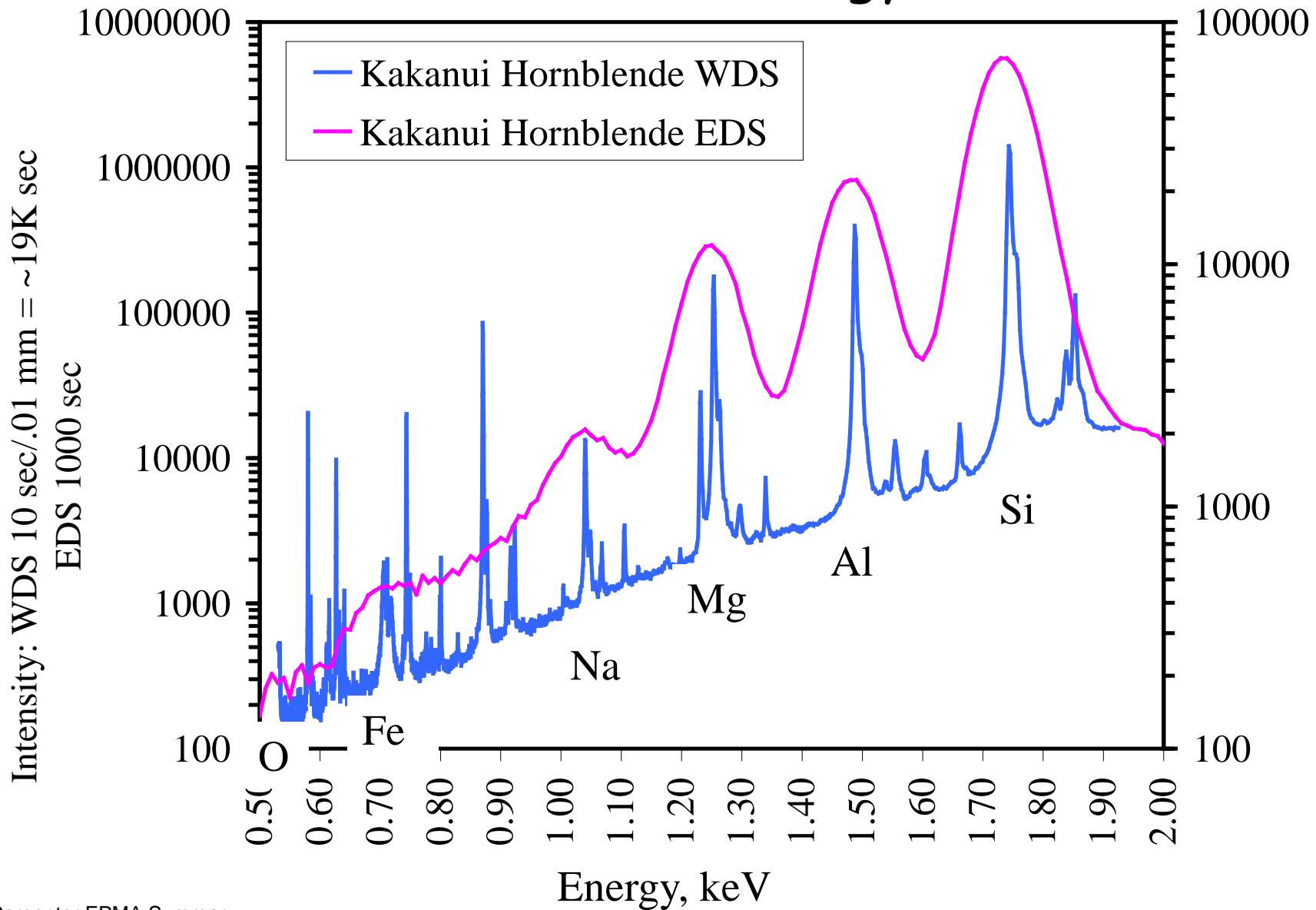
TAP Spec 2 25 KV 300 nA 70 - 266 mm 10 sec per point 0.1 / .09 mm step

WDS Scan TAP Kakanui Hornblende

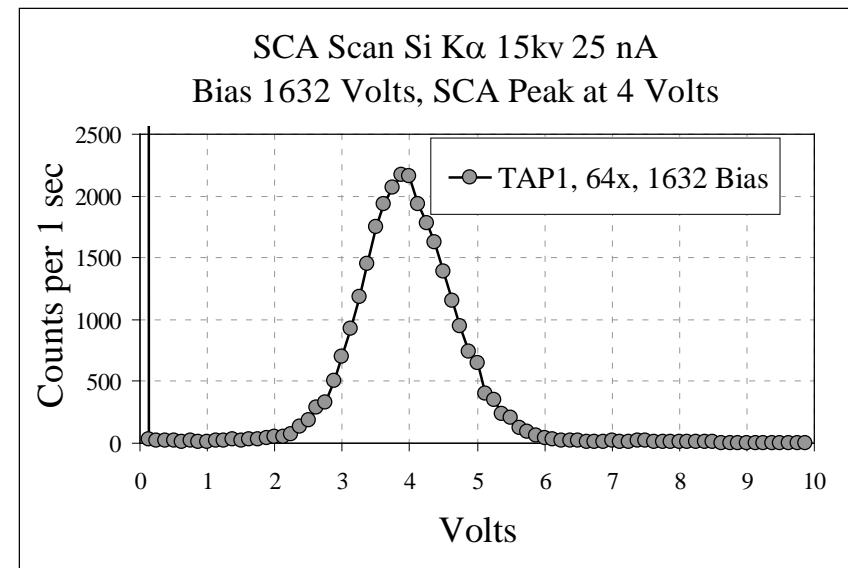
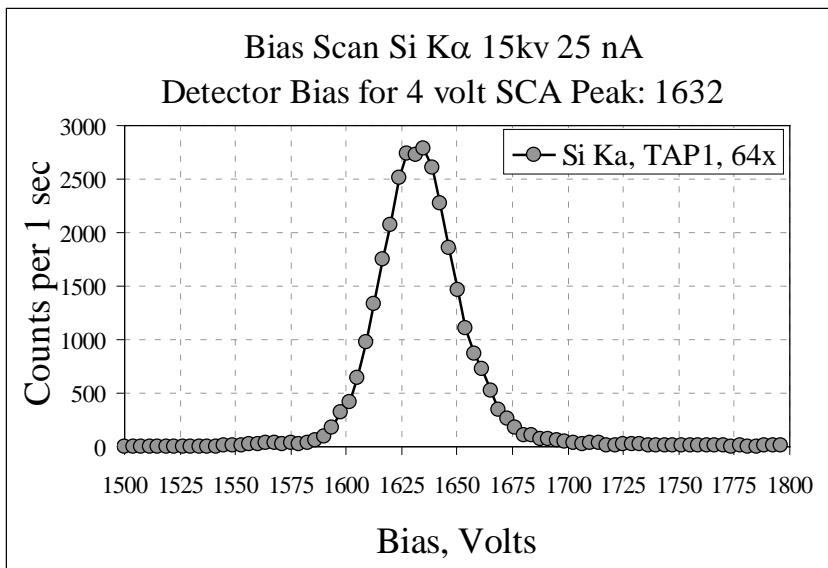


WDS and EDS Comparison: Kakanui Hornblende

WDS TAP GFC, SiLi EDS, Energy Scale



Pulse-height Analysis: WDS

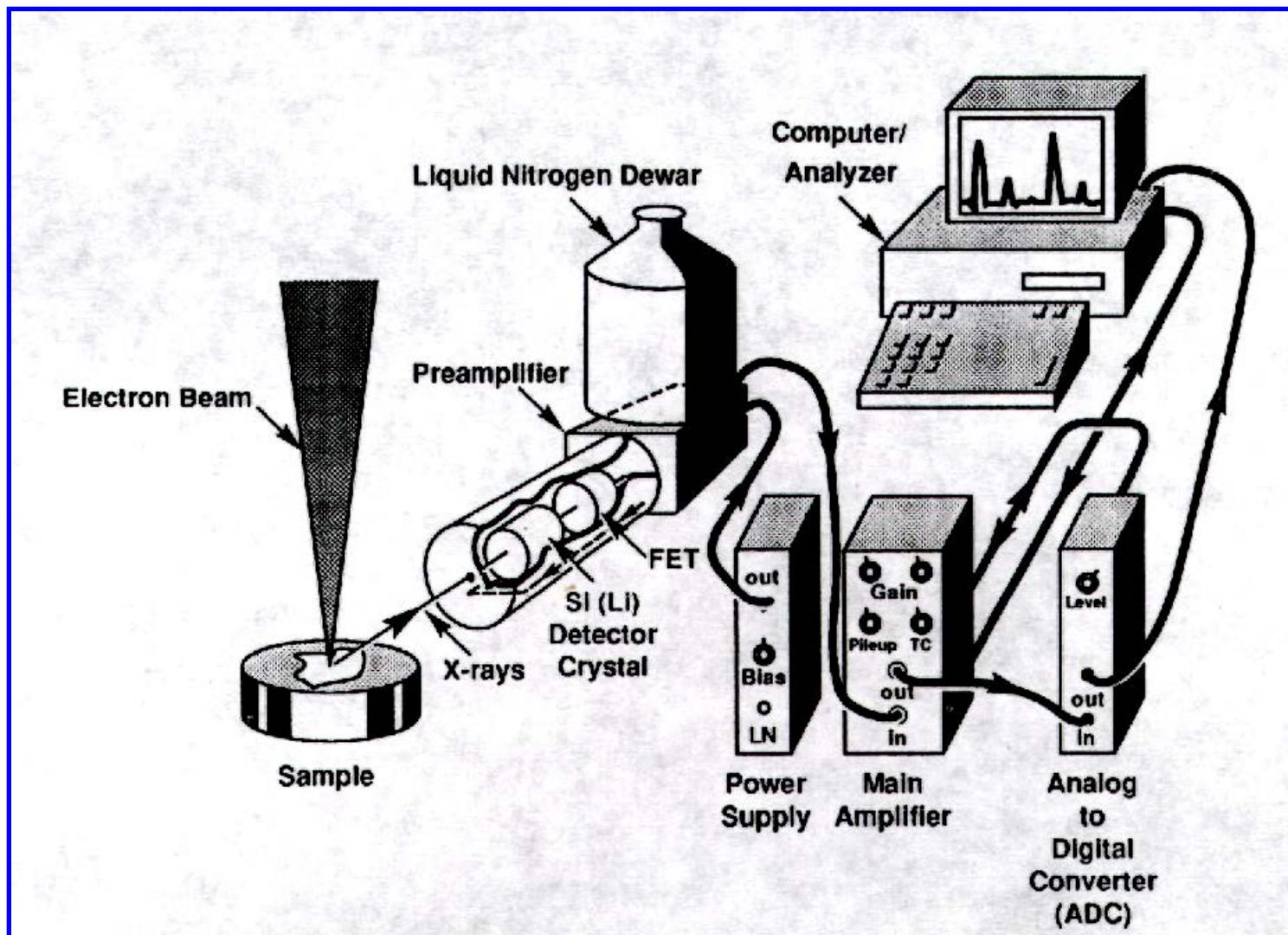


Detector bias scan for Si K α using 3.9 volt baseline and 0.2 volt window. The bias scan is used to set the detector bias necessary to achieve a 4 volt pulse on an SCA scan (from 0-10 Volts). Bias scans are typically performed in the 1500-1800 Volt range, which is the plateau region of the x-ray detector. This is for uniform application of detector gain and consistent detector deadtime behavior.

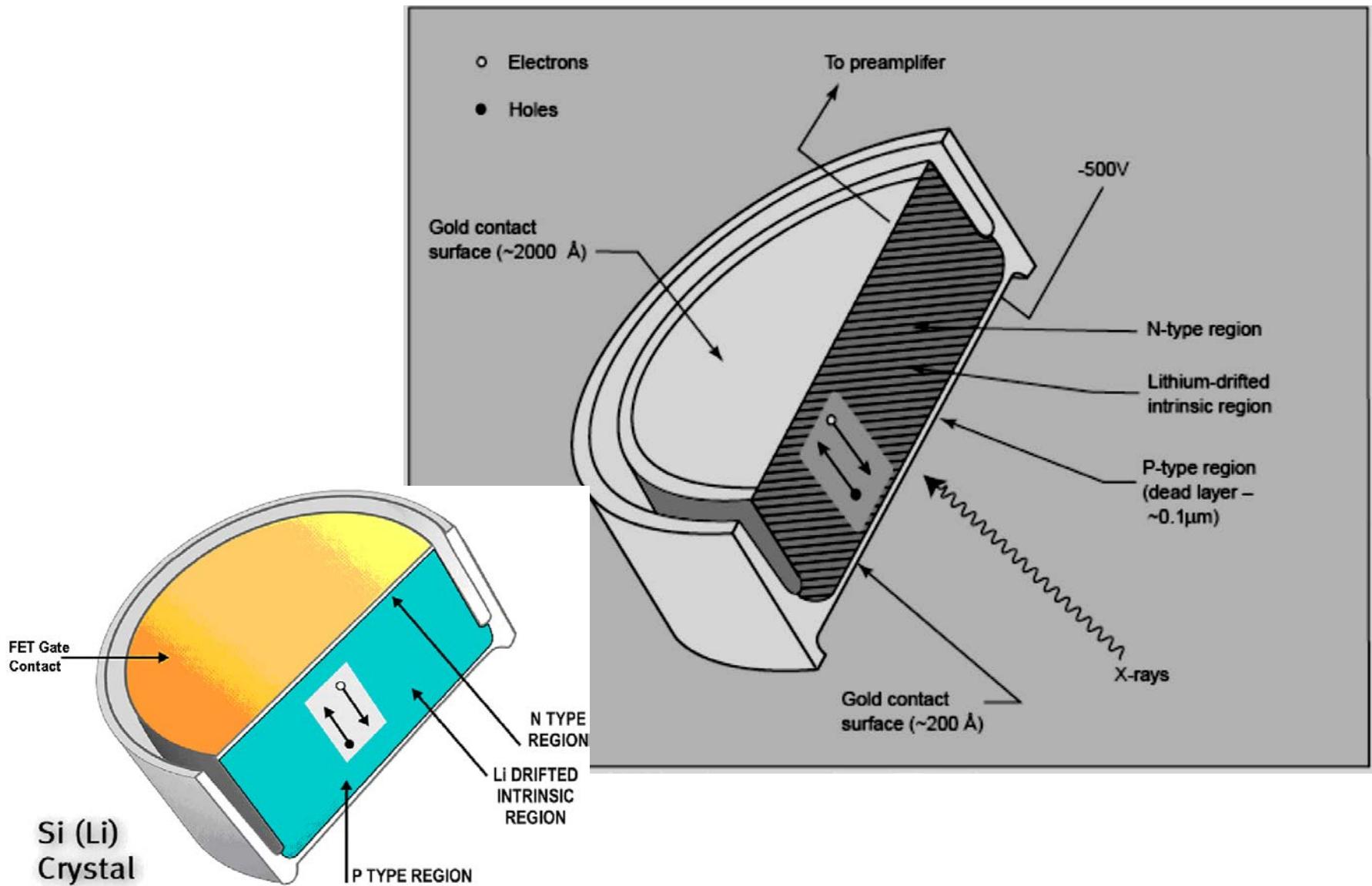
SCA (single channel analyzer) scan over detector voltages produced by entry of Si K α photons into the gas-flow counter when the WDS is set to the Si K α peak position. The detector bias is set to the value determined in the bias scan, which ensures that the SCA pulse is at the nominal value (here 4 Volts). An SCA scan is typically performed over the range 0-10 Volts. This uniform application of detector gain enables the use of uniform baseline and window values, and also minimizes the effect of gain variations on deadtime. Note the baseline noise below 0.25 Volts. At high count rates the pulse will shift to lower values and will eventually merge into the baseline due to inability of the counting electronics to resolve the pulse energy. For this reason it is important to monitor the bias and SCA voltage values used for an element.

EPMA Summary
Energy-Dispersive Spectrometry
Silicon Drift EDS Detectors
Cathodoluminescence

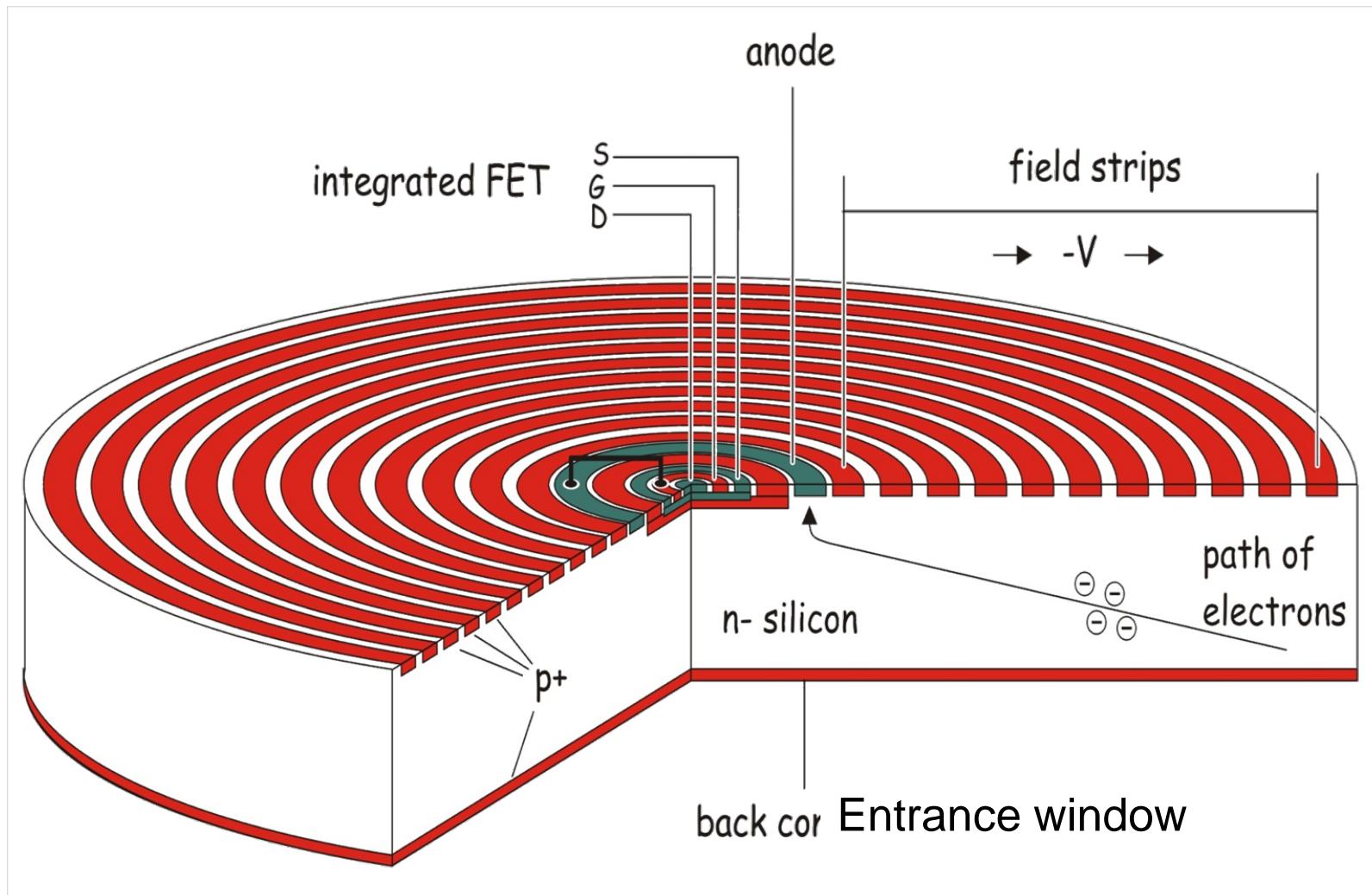
EDS Analyzer System



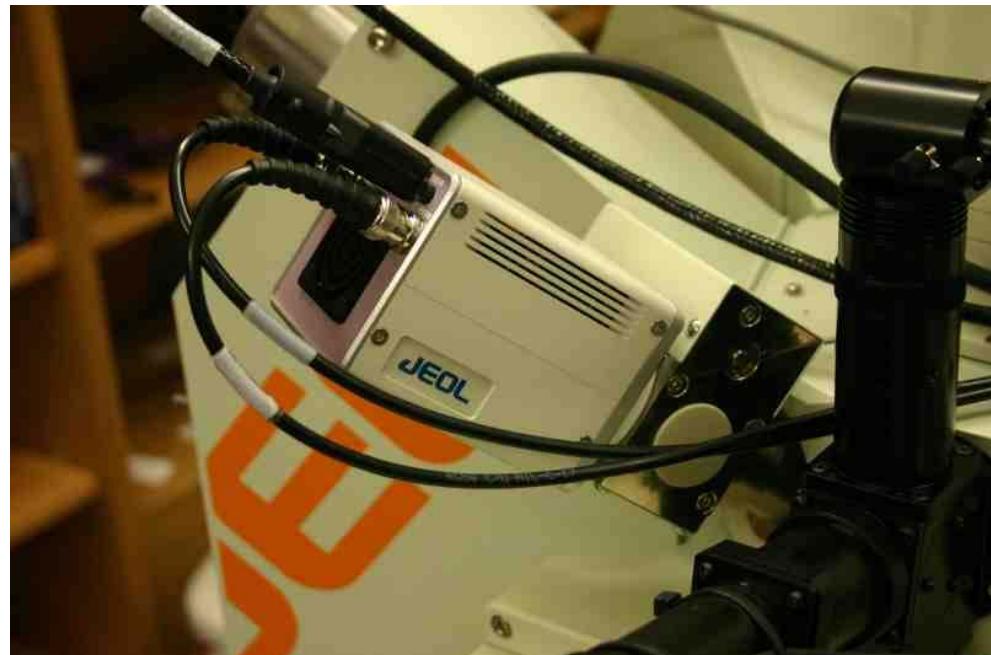
Conventional SiLi EDS Detector



Silicon Drift Detector

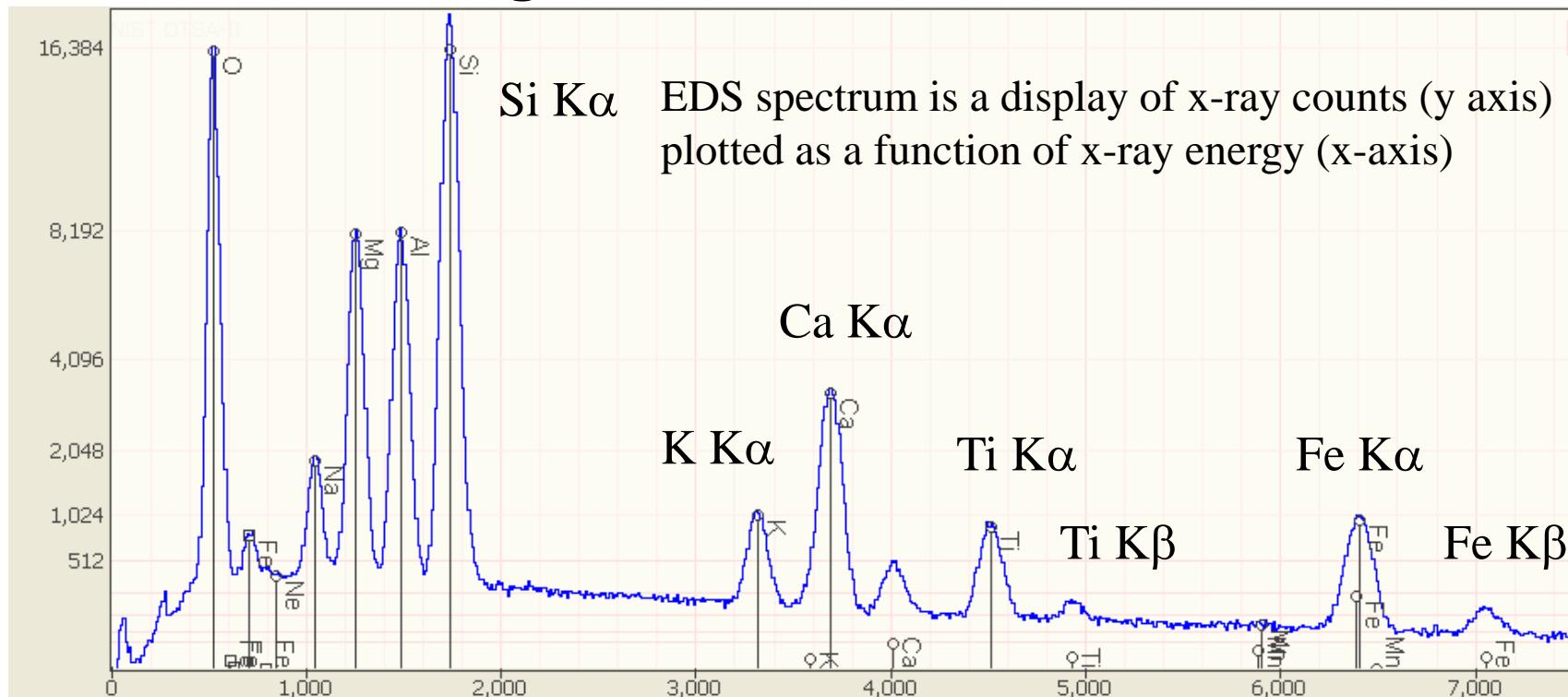


Washington University Earth and Planetary Sciences JEOL JXA-8200



JEOL e2v Silicon Drift Detector
130 eV resolution
3 time constants T3 T2 T1
Stage and beam mapping
Quantitative EDS analysis LLSQ

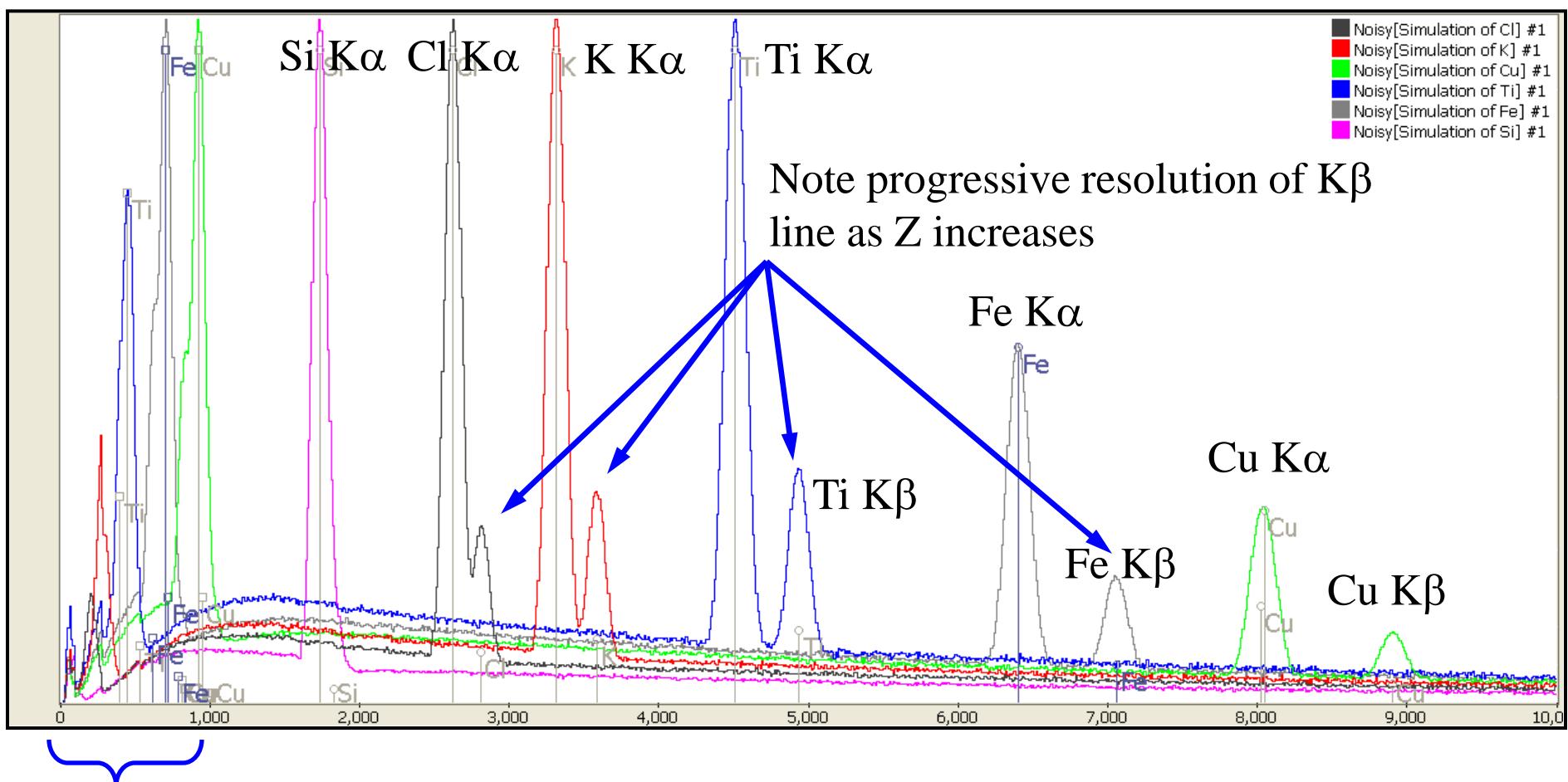
Kakanui Hornblende EDS X-ray Spectrum Generated Using DTSA-II



The EDS spectrum has **characteristic** x-ray peaks produced by specific transitions from the atom's excited state that are specific to each element. These characteristic peaks are superimposed on **continuum** ("Bremsstrahlung") produced by decelerating electrons. The EDS spectrum is used for routine identification of phases and for analysis. Note increasing energy resolution with increasing x-ray energy (Si vs. Fe) Peak overlaps (Na-Si K α peaks, and n-1 element K β peak on n element K α (K, Ca)

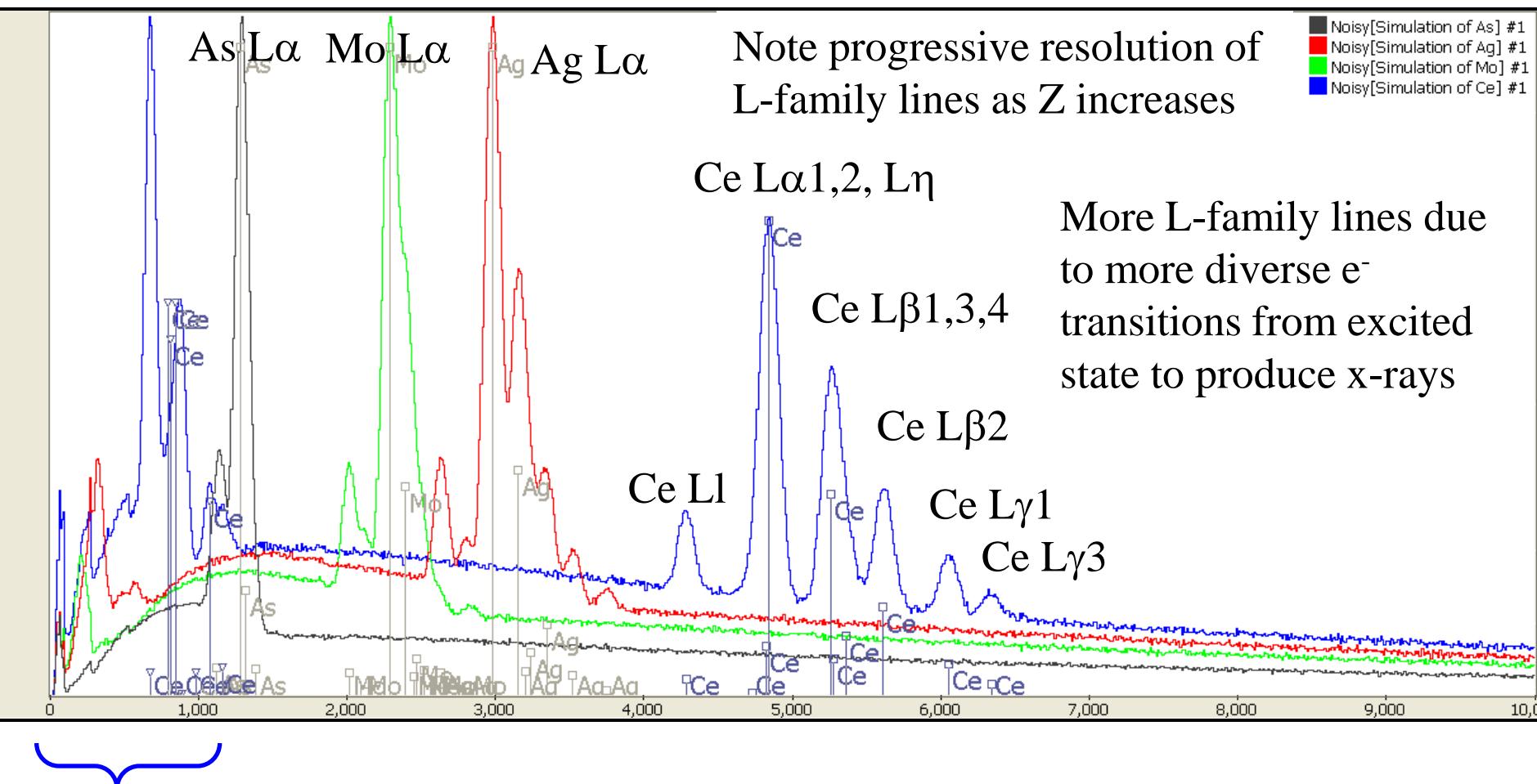
K-Family X-ray Spectra (EDS)

Generated Spectra Using DTSA-II, $E_0 = 15$ keV



Also note appearance of L-family lines as Z increases

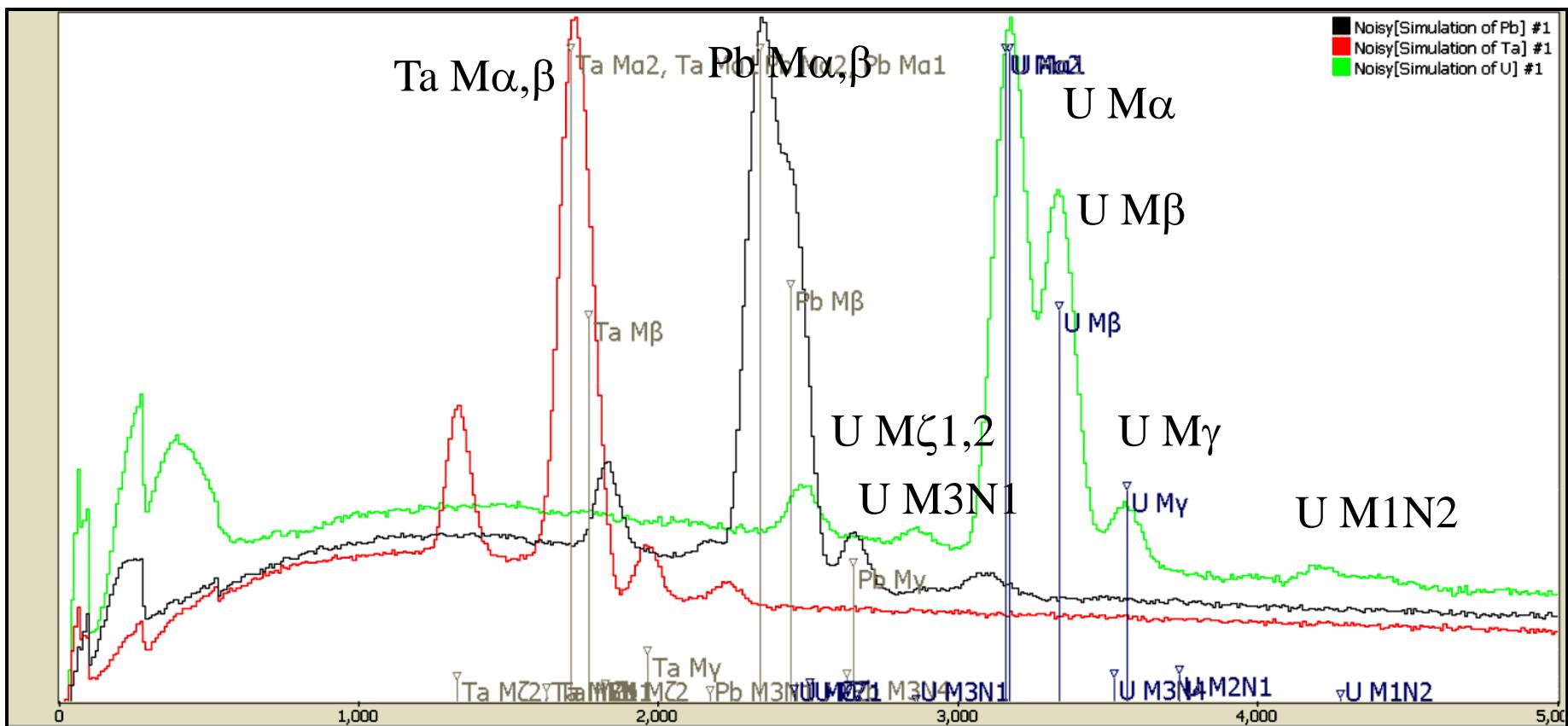
L-Family X-ray Spectra (EDS) Generated Using DTSA-II, $E_0 = 15$ keV



Also note appearance of M-family lines as Z increases

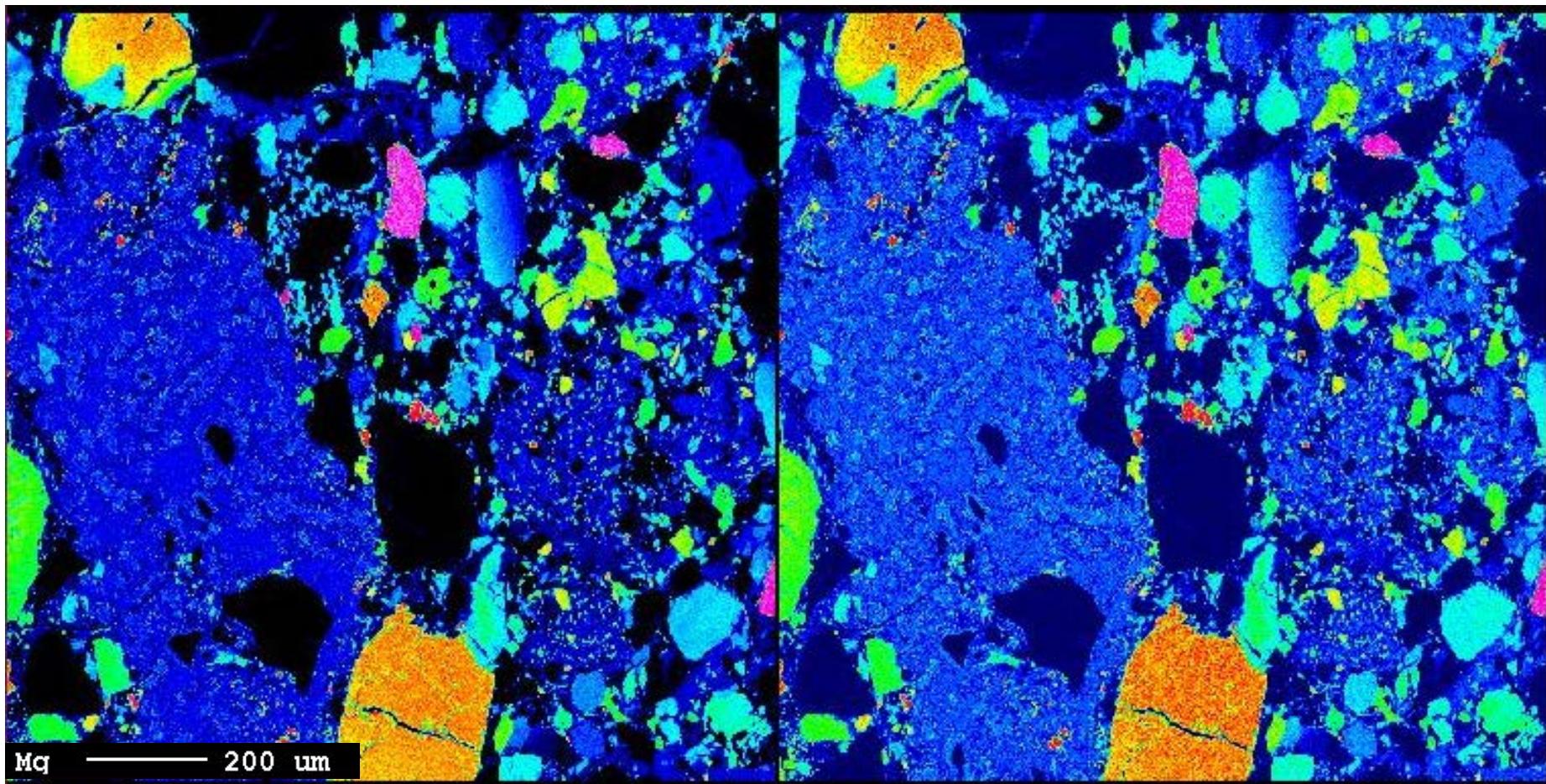
M-Family X-ray Spectra (EDS)

Generated Using DTSA-II, $E_0 = 15$ keV



Note progressive resolution of
M-family lines as Z increases

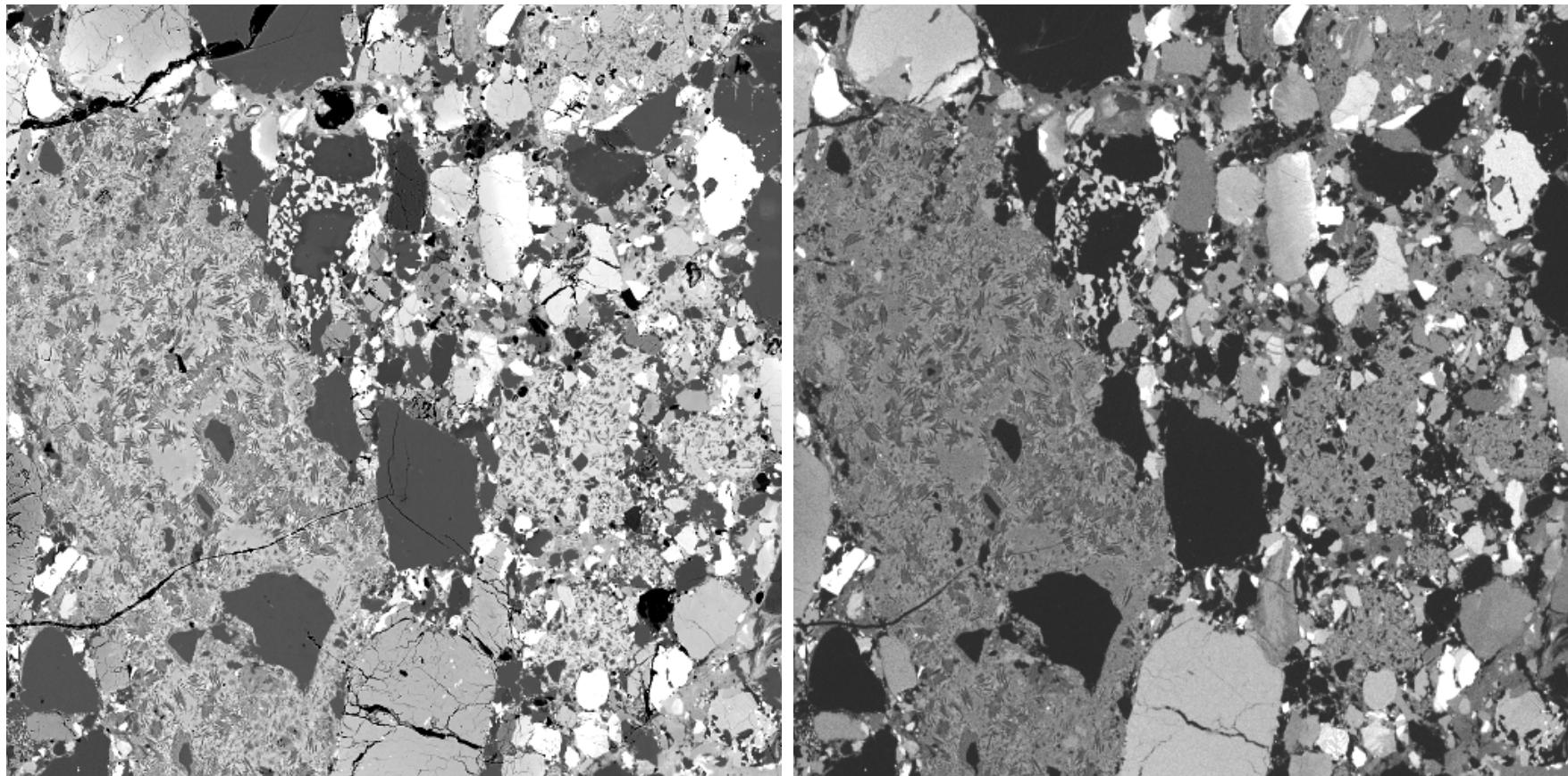
JEOL 8200 Stage Maps: Lunar Meteorite SAU169 WDS vs. SDD Mg @ 15KV, 120nA, 25ms dwell



Mg WDS 1061 max counts, Mg SDD 527 max counts
1024x1024 stage map, 8 hours

Lunar Meteorite SAU169

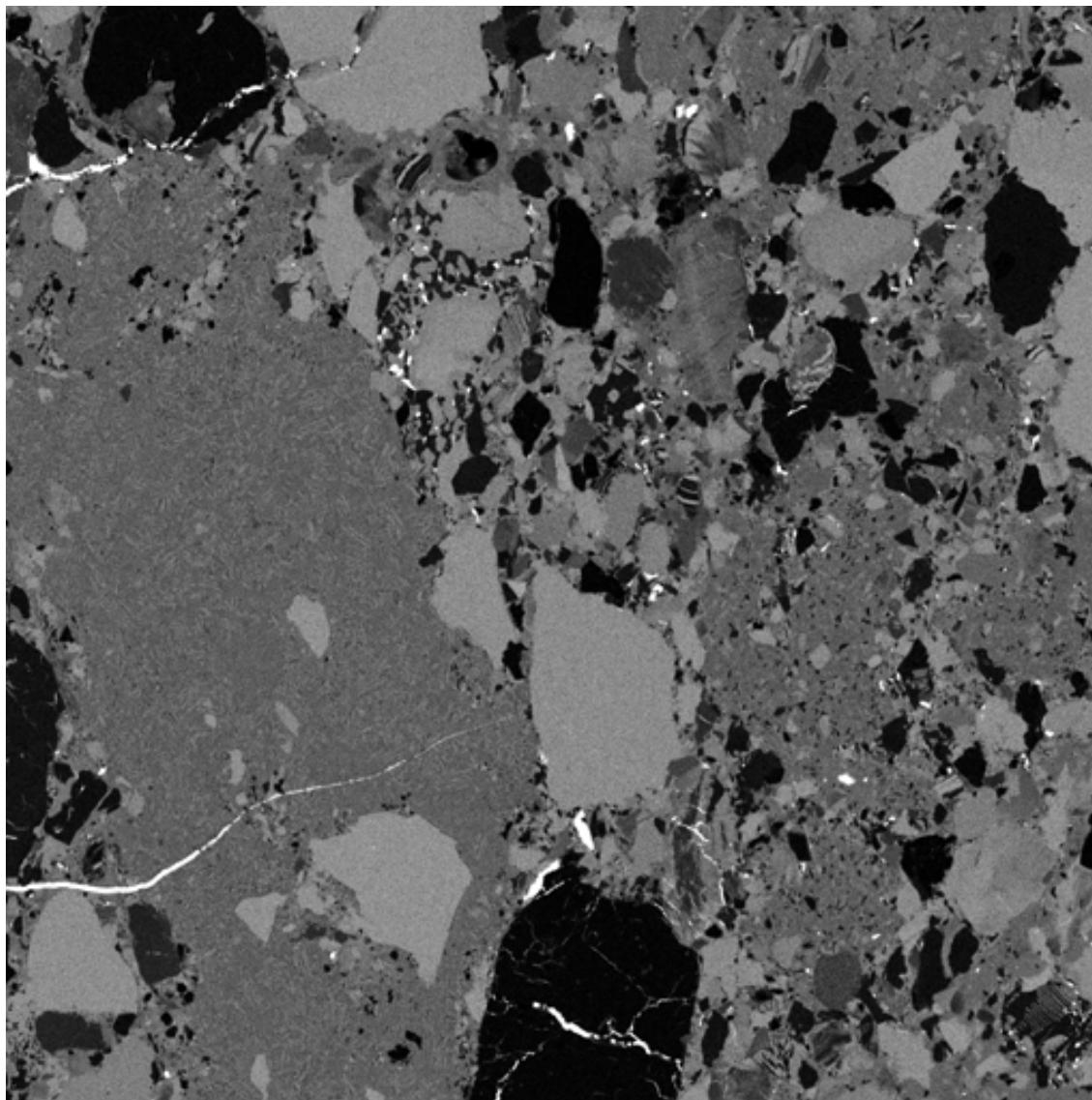
Stage map 1024x1024, 25 ms, 8 hr run



Backscattered electron vs. Fe SDD maps

Lunar Meteorite SAU169

Stage map 1024x1024, 25 ms, 8 hr run: Ca K α
SDD

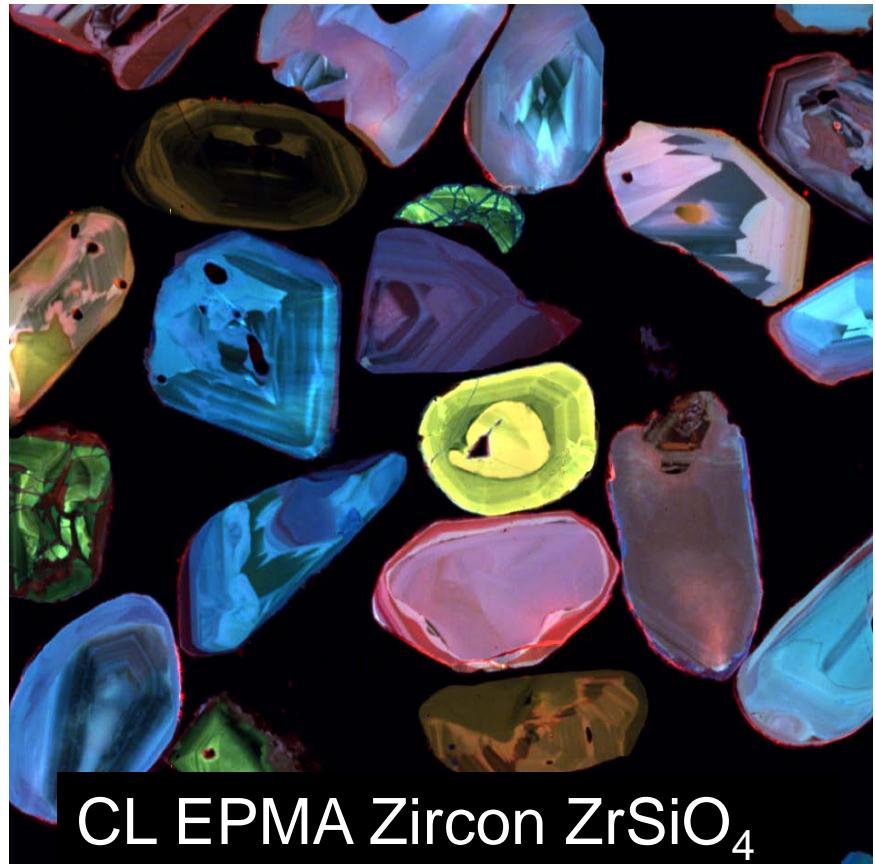


Cathodoluminescence Advances

Luminoscope CL image of typical single Crystal Diamond, South Africa



HFW ~3 mm (courtesy of S. Shirey)



Cathodoluminescence

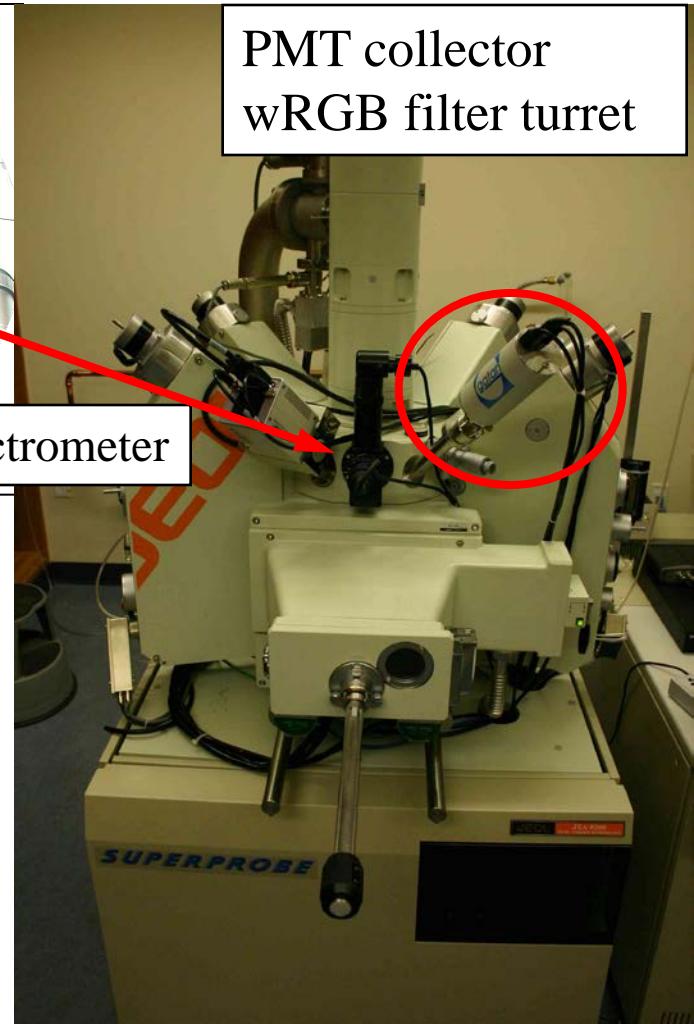
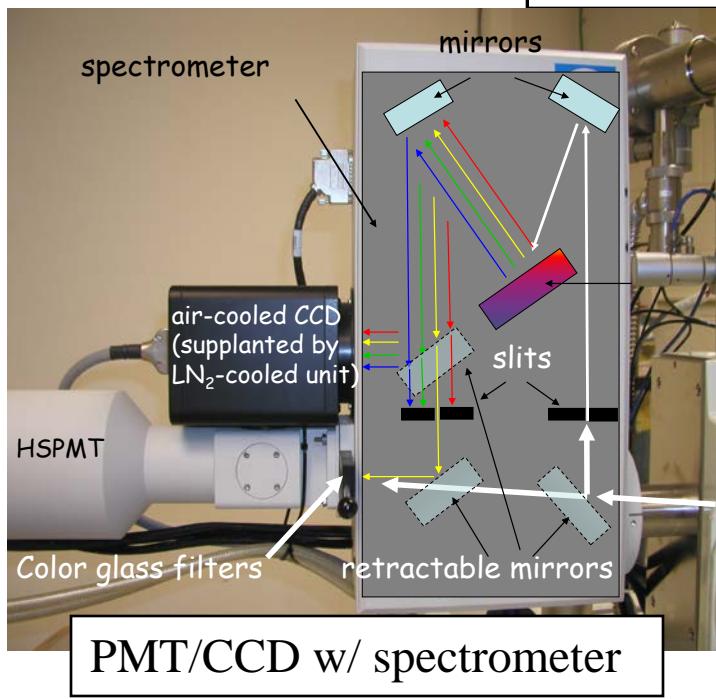
- ◆ CL is the emission of photons in the visible light region resulting from electron bombardment; EPMA with focused beam is excellent for CL
- ◆ The spectroscopy and interpretation of CL are complex
- ◆ CL is produced by defects (semiconductors) and/or trace elements (minerals)
- ◆ Activation can be produced by trace elements at ppb-ppm concentrations, well below nominal EPMA sensitivity (bse, x-ray)
- ◆ Minerals: activation by Mn, REE, etc., quenched at high concentration and by Fe
- ◆ CL collected by photomultiplier, parabolic mirror, fiber optic/grating systems
- ◆ Total CL signal vs. spectral and hyperspectral data collection
- ◆ Quantification requires spectrum peak deconvolution

Types of EPMA CL systems



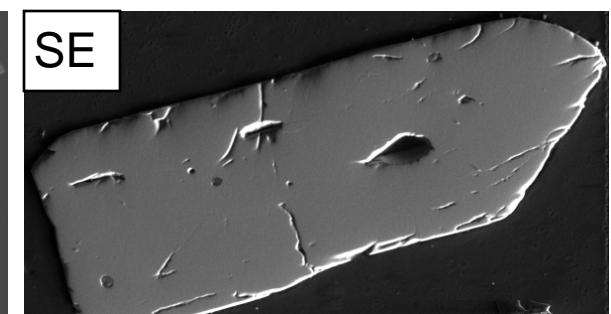
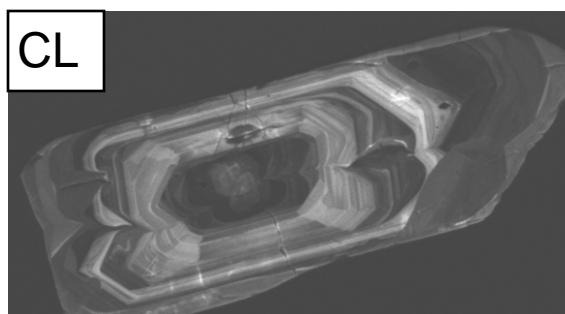
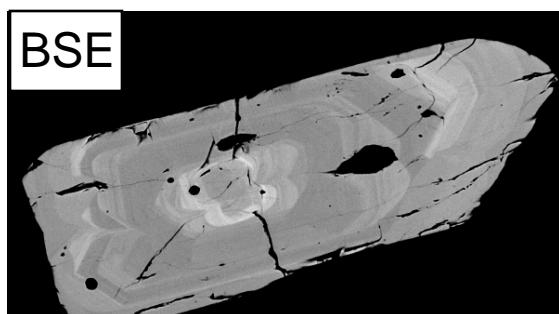
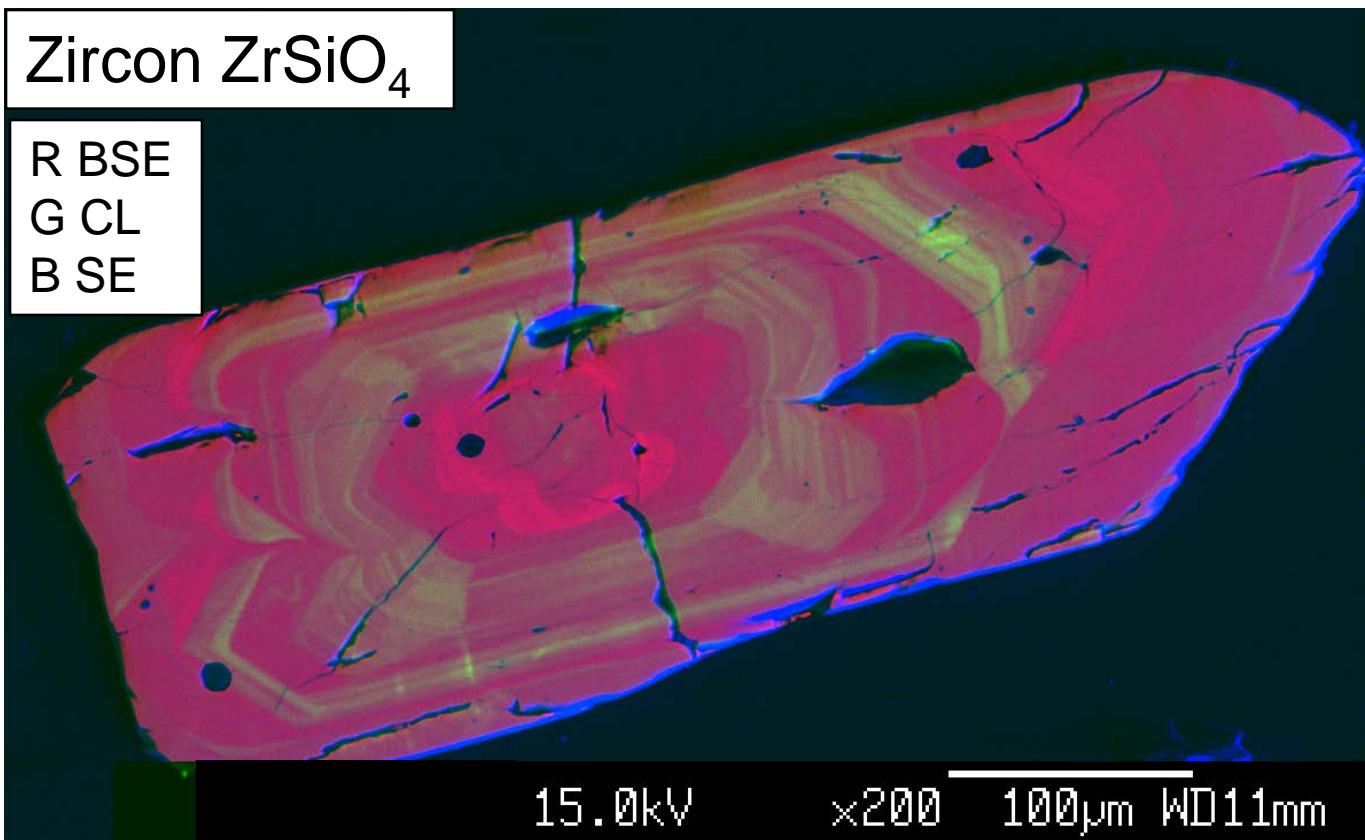
PMT collector
wRGB filter turret

Optical microscope w/ spectrometer



PMT/CCD w/ spectrometer

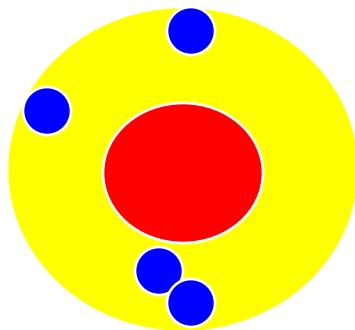
Combined CL, BSE, SE Images in Zircon



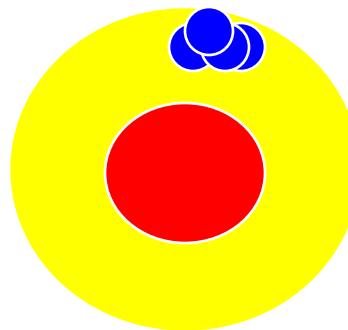
EPMA Summary
Precision and Accuracy
in
Microanalysis

Precision vs. Accuracy

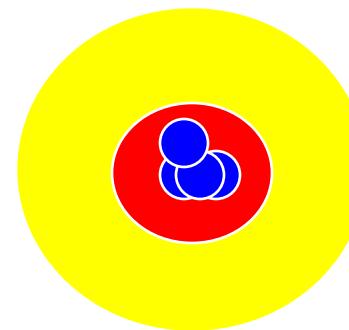
- ◆ Precision is the reproducibility of replicate measurements. High precision means reproducible. Target practice: grouping of impact points.
- ◆ Accuracy is the correctness of the measurement. Target practice: proximity to bullseye.



Poor precision
Poor accuracy



Good precision
Poor accuracy



Good precision
Good accuracy

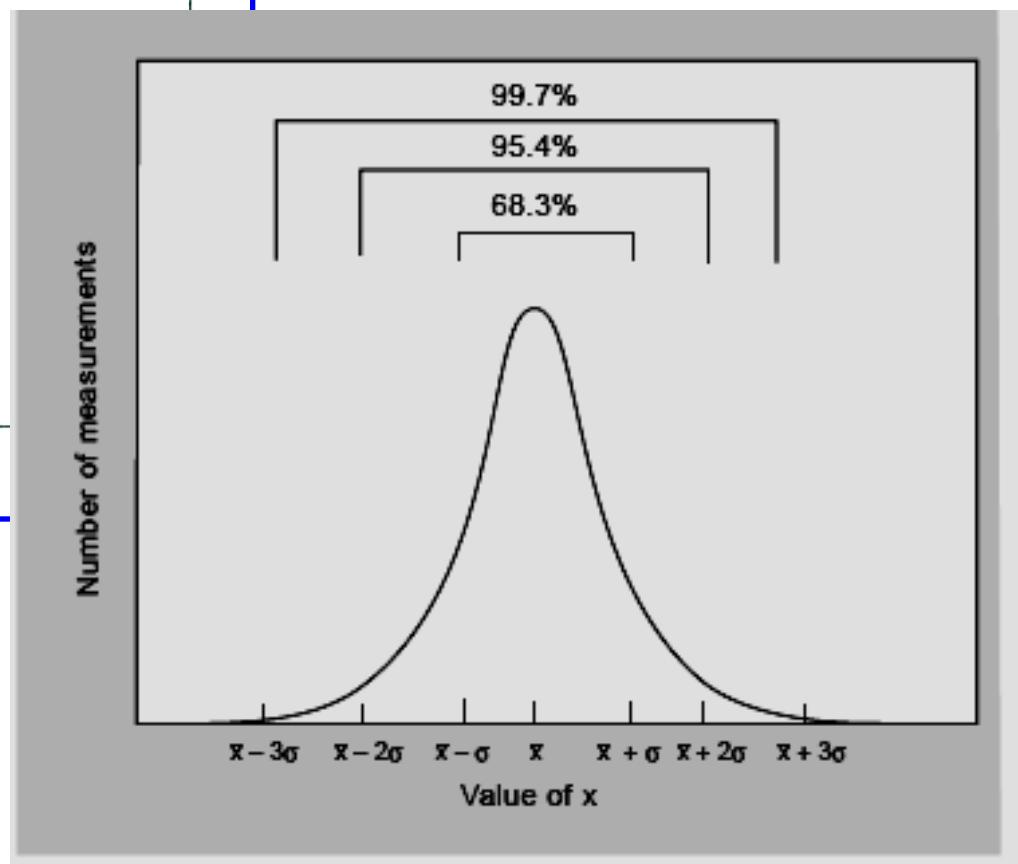
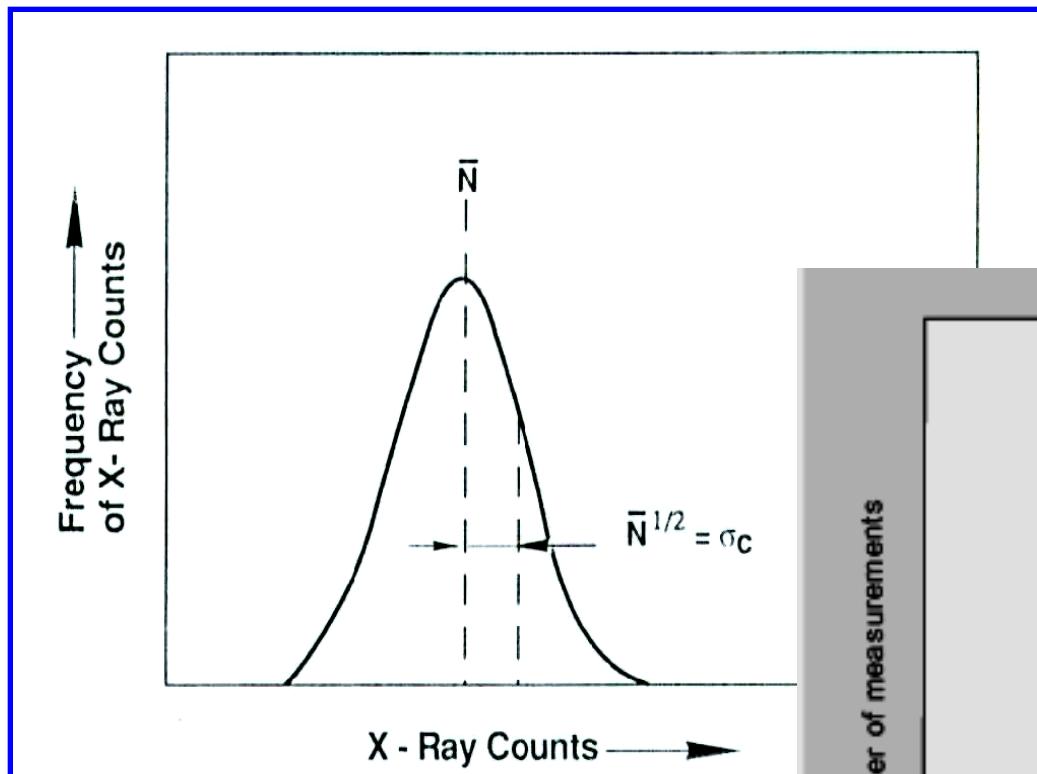
What is the Precision of Analysis?

- ◆ Precision is the reproducibility of a measurement, which accuracy is the degree of correctness of a measurement.
- ◆ X-ray production is statistical, and counting experiments yield a gaussian distribution of counts for which the standard deviation σ_c is the square root of the mean of the counts N_m :

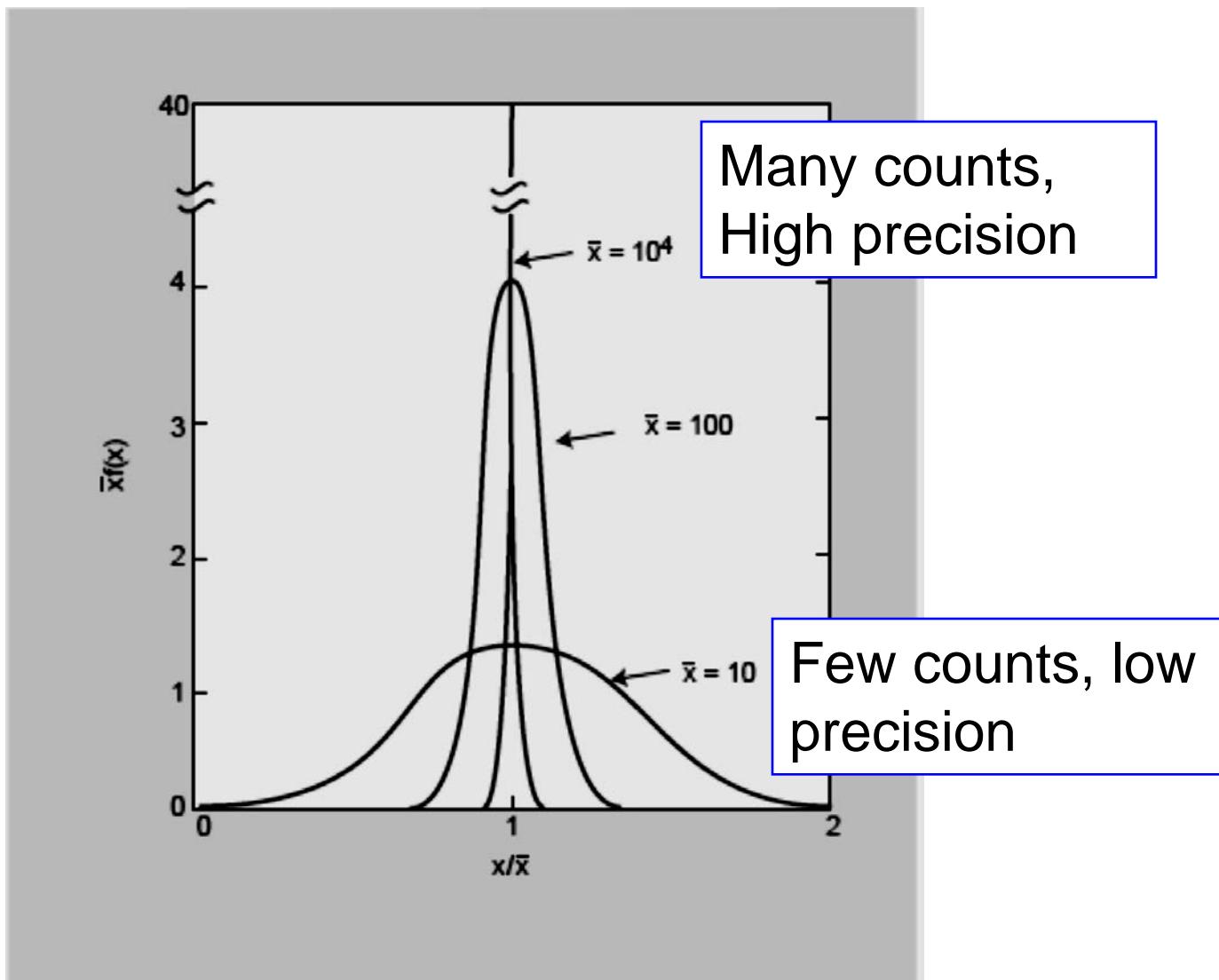
$$\sigma_c = (N_m)^{1/2}$$

- ◆ σ_c is 1-sigma (67% confidence level), multiply by 2 for 2-sigma (97%), by 3 for 3-sigma (99%).
- ◆ Without drift or other instrumental instabilities, the precision of the measurement is determined by the number of x-rays counted.
- ◆ In the case of EDS, the relatively poor P/B makes it unclear as to what is the “peak” and what is the “background”. Counts are obtained from the fitting procedure. Peak channel has highest P/B and is not the whole story.

Counting Statistics—Gaussian Distribution



Relative Precision as a Function of Number of X-rays Counted



Precision of X-ray Counting

Counts N_m	σ_c	$\sigma_c / N_m * 100$ (relative percent)
100	10	10 %
1000	31.62	3.16 %
10,000	100	1.0 %
100,000	316.2	0.316 %
1,000,000	1000	0.1 %

Actual Standard Deviation

- ◆ The actual standard deviation of a set of measurements is the sum of the standard deviation due to counting statistics alone, plus the actual variation of an element in a sample, plus any other factors that cause variation (drift, for example):

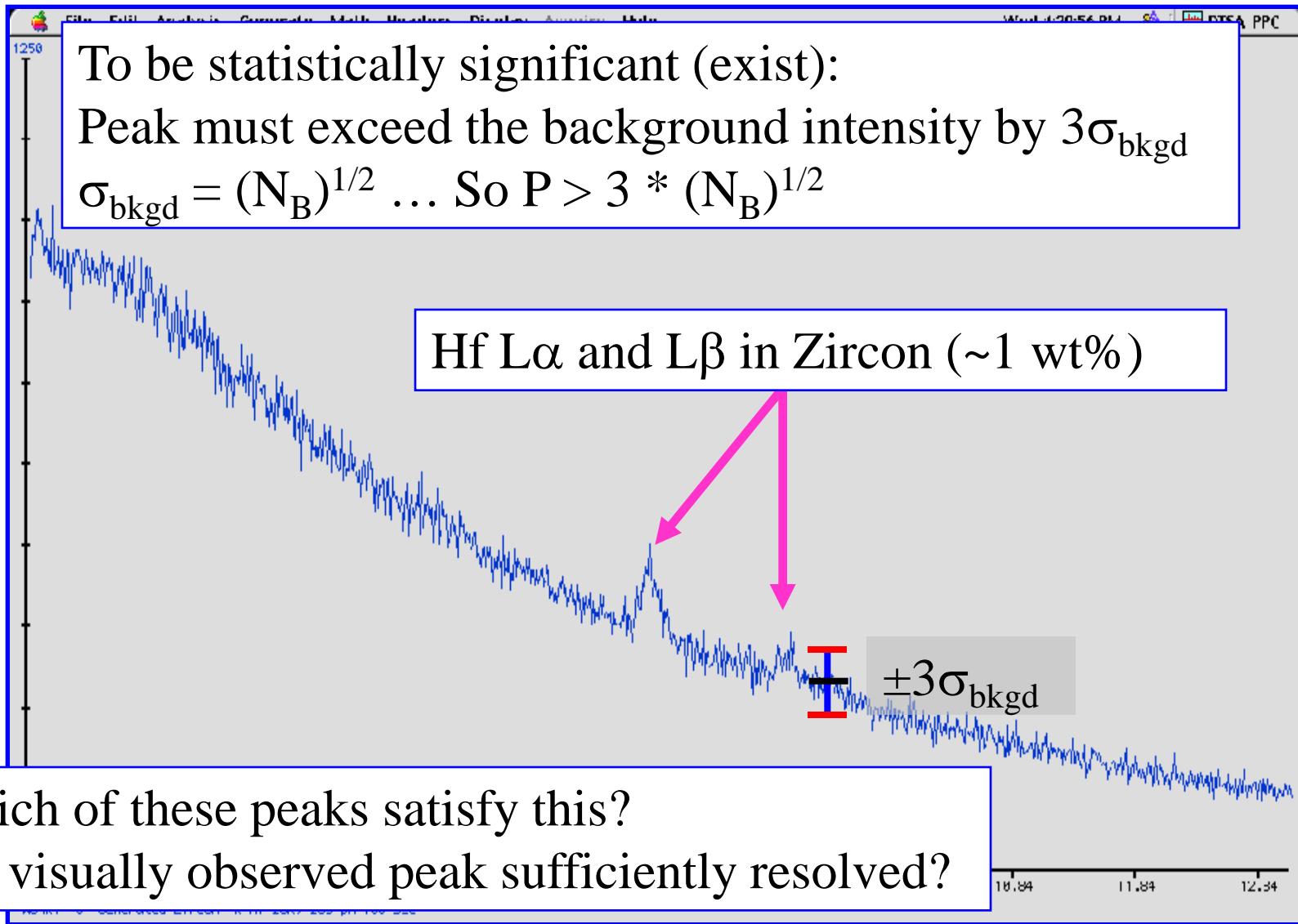
$$\sigma_{\text{Total}} = \sigma_{\text{count stats}} + \sigma_{\text{actual}} + \sigma_{\text{drift}} + \dots$$

- ◆ The actual standard deviation is:

$$S_c = \left[\sum_{i=1}^n \frac{(N_i - \bar{N}_i)^2}{(n-1)} \right]^{1/2}$$

where n is the number of measurements and N bar is the mean.

Are These Peaks Real?



Detection Limit

- ◆ An element is present when the peak exceeds 3 times the standard deviation of the background intensity measured at the peak position. This is a statistical approach, 3σ is 99% of the background variation and any counts outside of that range must be a “peak”.

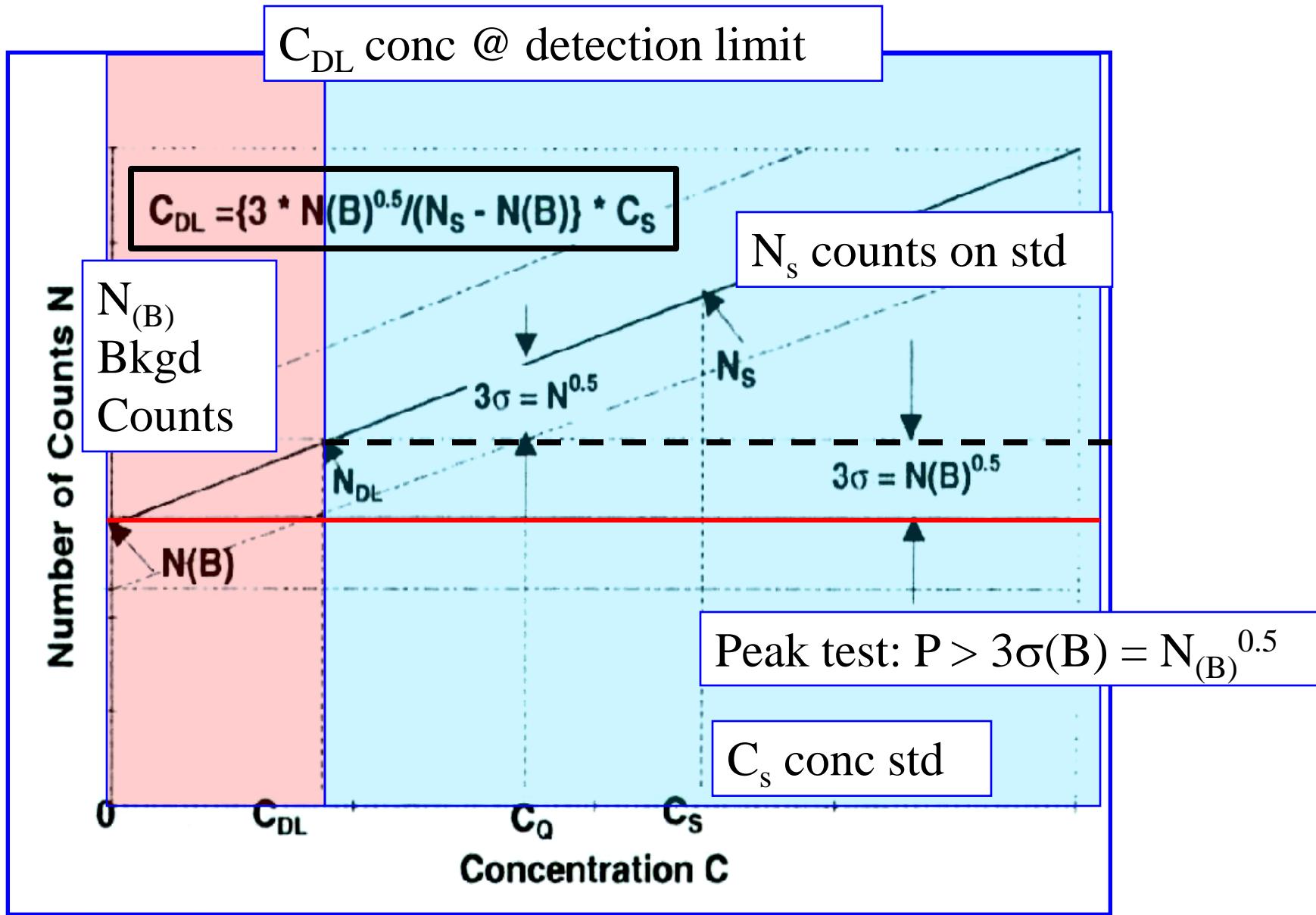
$$N_p > 3(N_b)^{1/2}, \text{ where } N_p \text{ is peak, } N_b \text{ is background}$$

- ◆ A convenient formula for calculation of detection limit is that of Ziebold (1967):

$$C_{DL} \geq \frac{3.29a}{(ntP \cdot P/B)^{1/2}}$$

where t is the time of measurement, n is # of repetitions, P is count rate on pure element, P/B is peak-to-background of pure element, and a is the c/k factor (i.e. ZAF). P²/B is the figure of merit.

Detection Limit



Detection Limit EDS vs. WDS

Table 9.17. Comparison of the Minimum Detectability Limit of Various Elements Using an EDS and WDS Detection System on the Basis of Optimized Operating Conditions^a

Analysis	Element	P (cps)	B (cps)	P/B	Wet chem. (wt %)	C _{DL} (wt %)
EDS	Na <i>K</i> α	32.2	11.5	2.8	3.97	0.195
	Mg <i>K</i> α	111.6	17.3	6.4	7.30	0.102
	Al <i>K</i> α	103.9	18.2	5.7	4.67	0.069
	Si <i>K</i> α	623.5	27.3	22.8	26.69	0.072
	Ca <i>K</i> α	169.5	19.9	8.5	12.03	0.085
WDS	Na <i>K</i> α	549	6.6	83	3.97	0.021
	Mg <i>K</i> α	2183	8.9	135	7.30	0.012
	Al <i>K</i> α	2063	16.1	128	4.67	0.008
	Si <i>K</i> α	13390	37.0	362	26.69	0.009
	Ca <i>K</i> α	2415	8.2	295	12.03	0.009

^a Geller (1977). Analysis of DI-JD-35: EDS data collected at 2000 cps for 180 s, dead-time corrected (25%) (1.75 nA probe current at 15 keV); WDS data collected for 30 s for each element; 180 s total analysis time (30 nA probe current at 15 keV).

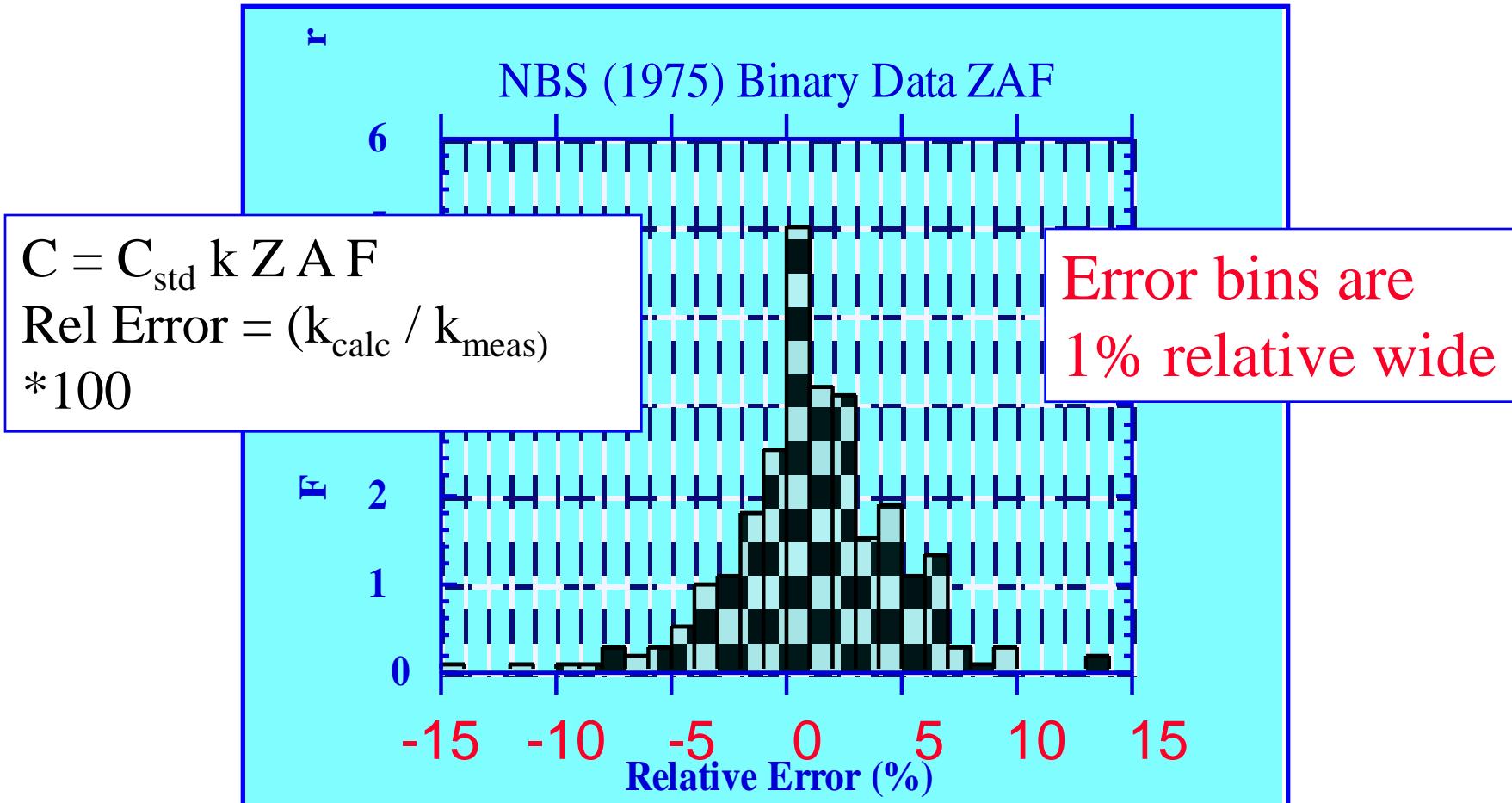
What is the Accuracy of Microanalysis?

- ◆ Precision is determined by the total number of x-rays counted for a given element. In general, measurements are made to better precision than accuracy warrants.
- ◆ Do not confuse precision with accuracy: replication of measurements does not tell you anything about the accuracy.
- ◆ Accuracy is dependent on all aspects of measurement:
Standards—how well characterized are the compositions?
Measurement process—systematic errors (takeoff angle?) instrument stable? Same conditions for std and smp?
Peak fitting procedure—calibration differences sample vs. standard, other artifacts of peak stripping?
X-ray correction algorithms and data sets—how accurate are the calculated results given “perfect” measurements?
- ◆ There is no global “accuracy” for EPMA measurements!

How Can Accuracy Be Determined?

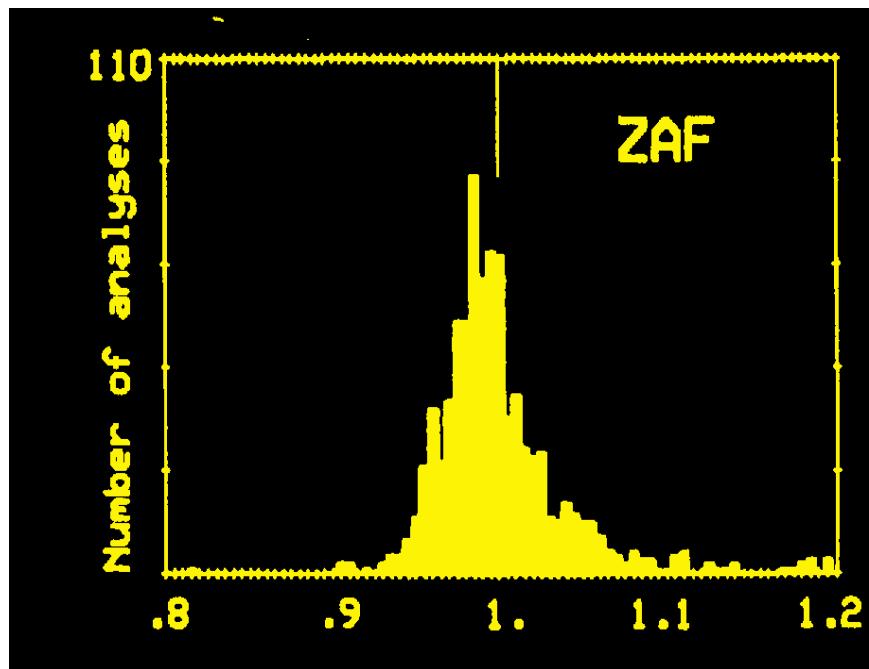
- ◆ Analysis of a secondary standard as an unknown, under the same conditions as sample measurement, allows determination of accuracy.
- ◆ Accuracy should be calculated from:
$$\% \text{Rel Acc} = 100 * (\text{Meas} - \text{Accepted}) / (\text{Accepted})$$
The sign is important + is high, - is low.
- ◆ Evaluation of experimental binary alloy data sets using different x-ray correction algorithms yields $k_{\text{corr}} / k_{\text{exp}}$ histograms. But what if these data sets were used to develop the correction procedure? Is this a test?
- ◆ Critical evaluation of experimental data sets suggests that for the best measurements the accuracy is ~1-2%, and is probably limited by precision.
- ◆ For “problem” elements the accuracy may be much worse, perhaps 5-10-20-50-100% !! Depends on concentration too.

Error Distribution for Matrix Corrections with ZAF/Standards (Pure Elements, Binary Compounds)

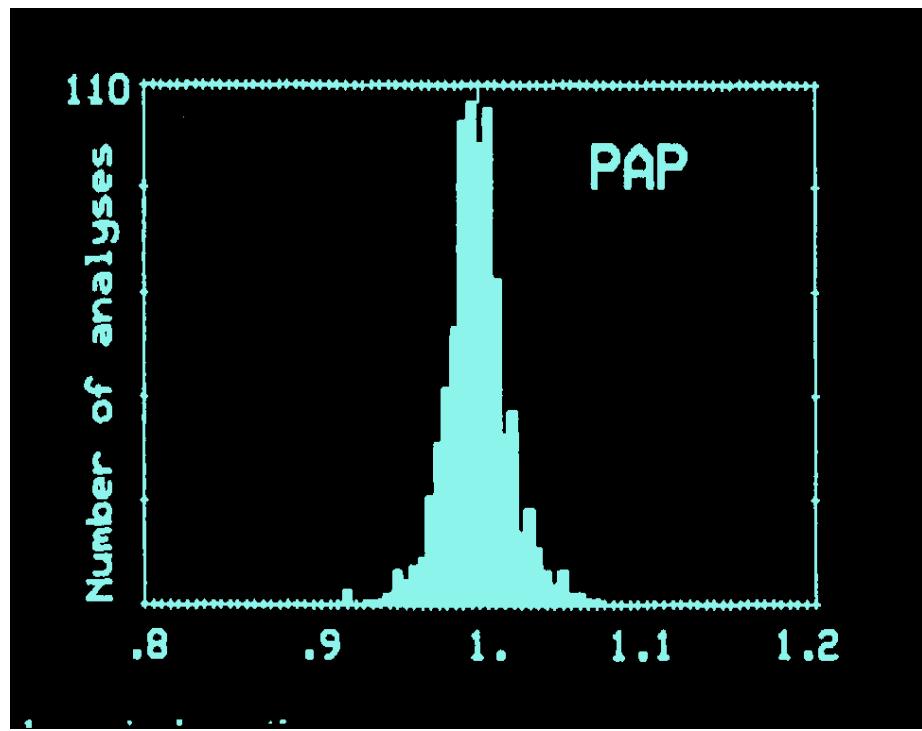


This distribution has $\sigma = 2.5\%$. 95% of analyses are expected to lie within $\pm 5\%$ relative of the correct value.

What If the $\Phi(\rho Z)$ /standards Method Is Used Instead of ZAF/standards?



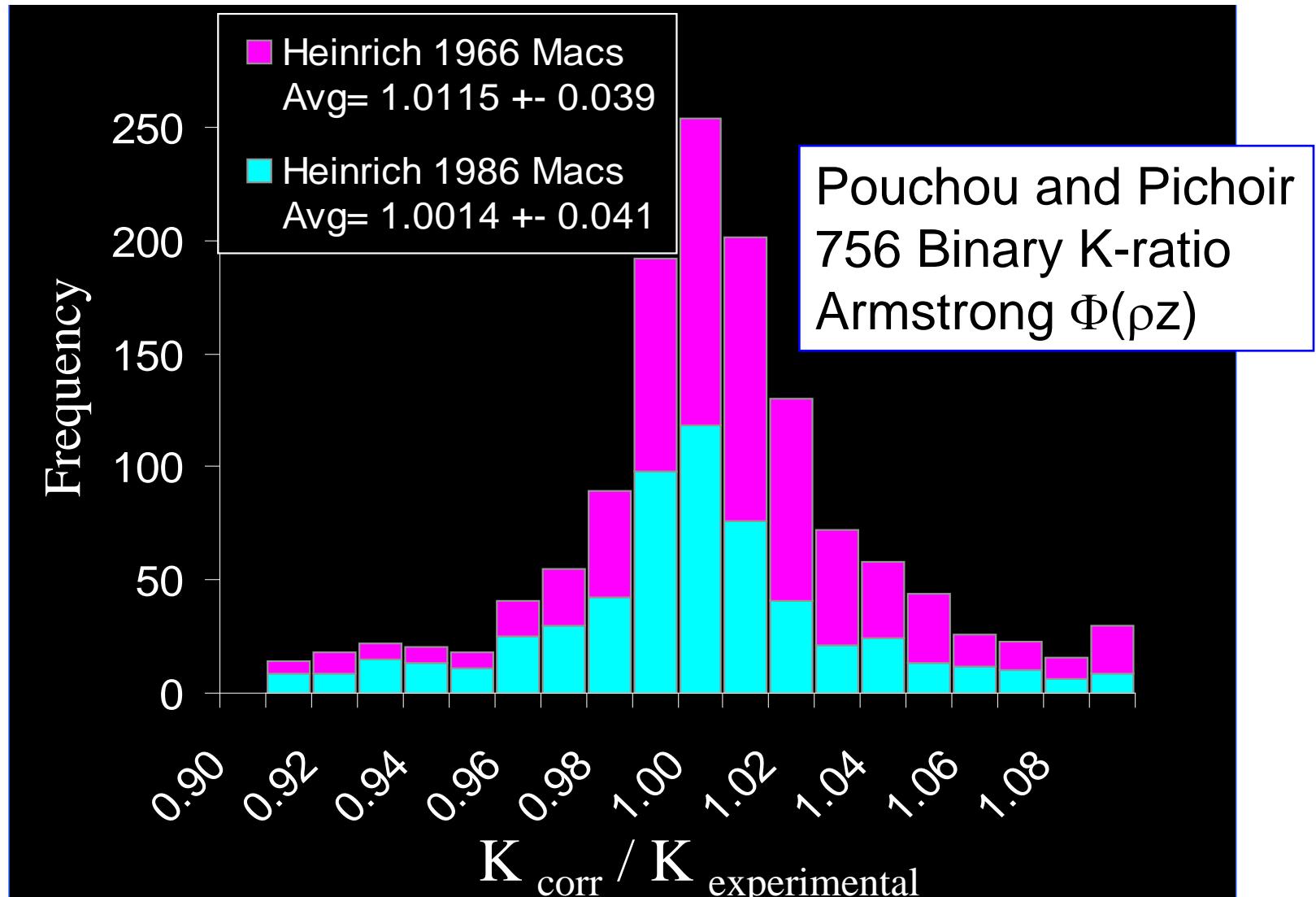
Conventional ZAF



Pouchou & Pichoir $\Phi(\rho z)$

The relative error distribution may be narrower yet!
The improvement depends on the particular elements.
Light element quantitation is particularly improved.

Heinrich 1966 vs. 1986 Mass Absorption Coefficients



EPMA Summary Resources

Books on EPMA and Quantitative Analysis

- ◆ Scanning Electron Microscopy and X-ray Microanalysis, 3rd Ed., Goldstein et. al. 2003 Plenum. Mandatory! – Previous editions very good
- ◆ Electron Probe Quantitation, Heinrich and Newbury. 1991 Plenum. “The green book”.
- ◆ Electron Microprobe Analysis, 2nd Ed., Reed. 1997 Cambridge. Very good.
- ◆ Electron Microprobe Analysis and Scanning Electron Microscopy in Geology, 2nd Ed. Reed. 2006 Cambridge.
- ◆ X-ray Spectrometry in Electron Beam Instruments, Williams, Goldstein, and Newbury. 1995 Plenum.
- ◆ Quantitative Electron-Probe Microanalysis, Scott and Love, 2nd Ed., 1995 Prentice Hall.

Websites with EPMA Information

- ◆ Paul Carpenter's lab web site
<http://xraysrv.wustl.edu/>
- ◆ John Fournelle's lab web site (see course materials)
<http://www.geology.wisc.edu/~johnf/sx51.html>
- ◆ John Wittke's lab web site
<http://www4.nau.edu/microanalysis/Microprobe/Probe.html>

Microanalysis Software

- ◆ Casino – Monte Carlo simulation program for electron scattering and x-ray $\Phi(\rho z)$ distributions.
<http://www.gel.usherbrooke.ca/casino>
- ◆ CalcZAF – A collection of ZAF and $\Phi(\rho z)$ algorithms used to convert relative x-ray intensities to concentration. Based on CITZAF (Armstrong).
<http://epmalab.uoregon.edu/calczaf.htm>
- ◆ TRYZAF – Version of CITZAF used to compare correction algorithms and data sets for given elements.
<http://www.cstl.nist.gov/div837/Division/outputs/software.htm>
- ◆ GMR – $\Phi(\rho z)$ thin-film correction program, includes continuum fluorescence correction, used to calculate K or C for multilayer thin films on substrates.