

Performance of the r²SCAN functional in transition metal oxides

S. Swathilakshmi, Reshma Devi, and Gopalakrishnan Sai Gautam*

Department of Materials Engineering, Indian Institute of Science, Bengaluru, 560012, India

E-mail: saigautamg@iisc.ac.in

Crystal Structures

The crystal structures of all transition metal oxides (TMOs) considered in this work are displayed in Figure S1. CrO₂, MnO₂ and TiO₂ have a rutile (*P*4₂/*mnm*) ground state structure. In rutile, the metal ion forms a body-centred tetragonal lattice with six anions forming an octahedron around each metal ion while each anion is coordinated to three cations. These octahedra share edges to form chains along the *c*-axis, with distinct chains connected by shared corners.^{1–5} TiO₂, CrO₂, and MnO₂ are diamagnetic, metallic ferromagnetic (FM),³ and antiferromagnetic, (AFM)⁶ respectively. VO₂, which exhibits a rutile structure at high temperatures, undergoes a transition to a monoclinic (*P*2₁/*c*) symmetry at temperatures below 341 K.⁷ This structural transition is due to a metal-to-insulator transition (MIT), which causes the distance between V⁴⁺ to decrease, forming homopolar bonds, and trapping the *d*-electrons.⁷ VO₂ exhibits a FM configuration.

Ti₂O₃ and Cr₂O₃ adopt a hexagonal (*R*3̄*cR*) ground state symmetry with oxygen in hexagonal close packing and metal ions occupying two-thirds of the octahedral sites.^{8,9} Cr₂O₃ exhibits a ↑↓↑↓ magnetic configuration while Ti₂O₃ displays a ↑↓↓↑ magnetic configuration on the transition metal (TM) centers along the *a*-axis.^{10,11} V₂O₃ undergoes structural dis-

tortion from room temperature hexagonal ($R\bar{3}c$) with paramagnetic configuration to an AFM monoclinic ($I2/a$) form below 160-170 K due to a MIT.¹²⁻¹⁴ The structural transition is associated with rotation of V-V bonds while the electrical transition is associated with an increase of V-V distance across the face shared octahedra. Fe_2O_3 exhibits hexagonal ($R\bar{3}c$) structure with oxygen forming a hexagonal close packed lattice and Fe^{3+} occupying two-thirds of the trigonally distorted octahedral sites. The AFM configuration of Fe_2O_3 is displayed in Figure S1.¹⁵

The rocksalt structures ($Fm\bar{3}m$) considered in this work include VO, MnO, FeO, CoO, and NiO, all of which adopt the type-II AFM configuration as their ground states.¹⁶⁻²¹ To model the type-II AFM configuration, we used a $2 \times 2 \times 2$ supercell of the rocksalt rhombohedral primitive cell in VO, MnO, and FeO, while we used a $1 \times 2 \times 1$ supercell of the primitive for NiO and CoO.

The spinels considered in this study include Mn_3O_4 , Fe_3O_4 , and Co_3O_4 . Among the spinels, Mn_3O_4 is tetragonally distorted ($I41/amd$) due to the Jahn-Teller distortion of the Mn^{3+} present in the octahedral sites of the structure.²² Also, we consider the “FIM6” ferrimagnetic configuration since it was reported to be the most stable among the 6 different ferrimagnetic configurations in our previous work.²³ At room temperature, Fe_3O_4 exhibits an inverse cubic spinel ($Fd\bar{3}m$) with O atoms forming cubic close packed array, Fe^{3+} occupying the tetrahedral sites, and both Fe^{3+} and Fe^{2+} randomly distributed in the octahedral sites. Additionally, Fe_3O_4 is reported to exhibit a Verway transition away from its room temperature cubic structure to either an orthorhombic²⁴ or a monoclinic²⁵ structure upon relaxation at low temperatures (< 120 K). This results in charge ordering of the Fe^{2+} and Fe^{3+} in the octahedral sites²⁶ and an opening of the band gap.²⁷ Co_3O_4 exhibits the primitive spinel structure ($Fd\bar{3}m$) with Co^{2+} occupying tetrahedral sites in high spin state in an AFM configuration while Co^{3+} occupy octahedral sites in their low spin, diamagnetic state.²⁸⁻³⁰

We consider FM-ordered $\alpha\text{-Mn}_2\text{O}_3$ with the bixbyite structure similar to our previous work,²³ wherein each Mn^{3+} ions are octahedrally coordinated with O while each O atom is

surrounded by 4 Mn atoms. Below 300 K, Mn_2O_3 exhibits Jahn-Teller distortion from cubic to an orthorhombic form ($Pbca$).³¹ LiNiO_2 possesses a layered structure with an alternating layer of Li and Ni atoms,³² and we used a primitive cell of LiNiO_2 ($P1m1$) in our calculations. V_2O_5 exhibits an orthorhombic ($Pmmn$) structure, and adopts a layered morphology with each oxygen coordinated vanadium (i.e., VO_5 square pyramids) sharing edges and corners.³³

CuO exhibits a monoclinic ($C2/c$) ground state with AFM configuration where each oxygen atom is surrounded by a distorted copper tetrahedron while each copper atom is surrounded by four oxygen in a square-planar configuration.³⁴ We used a $2 \times 1 \times 2$ supercell to model the AFM ground state of CuO . In Cu_2O , oxygen forms a BCC lattice with each oxygen ion surrounded by a tetrahedron of copper ions while copper is coordinated with two oxygens and thus exhibits a linear O-Cu-O configuration.^{34,35}

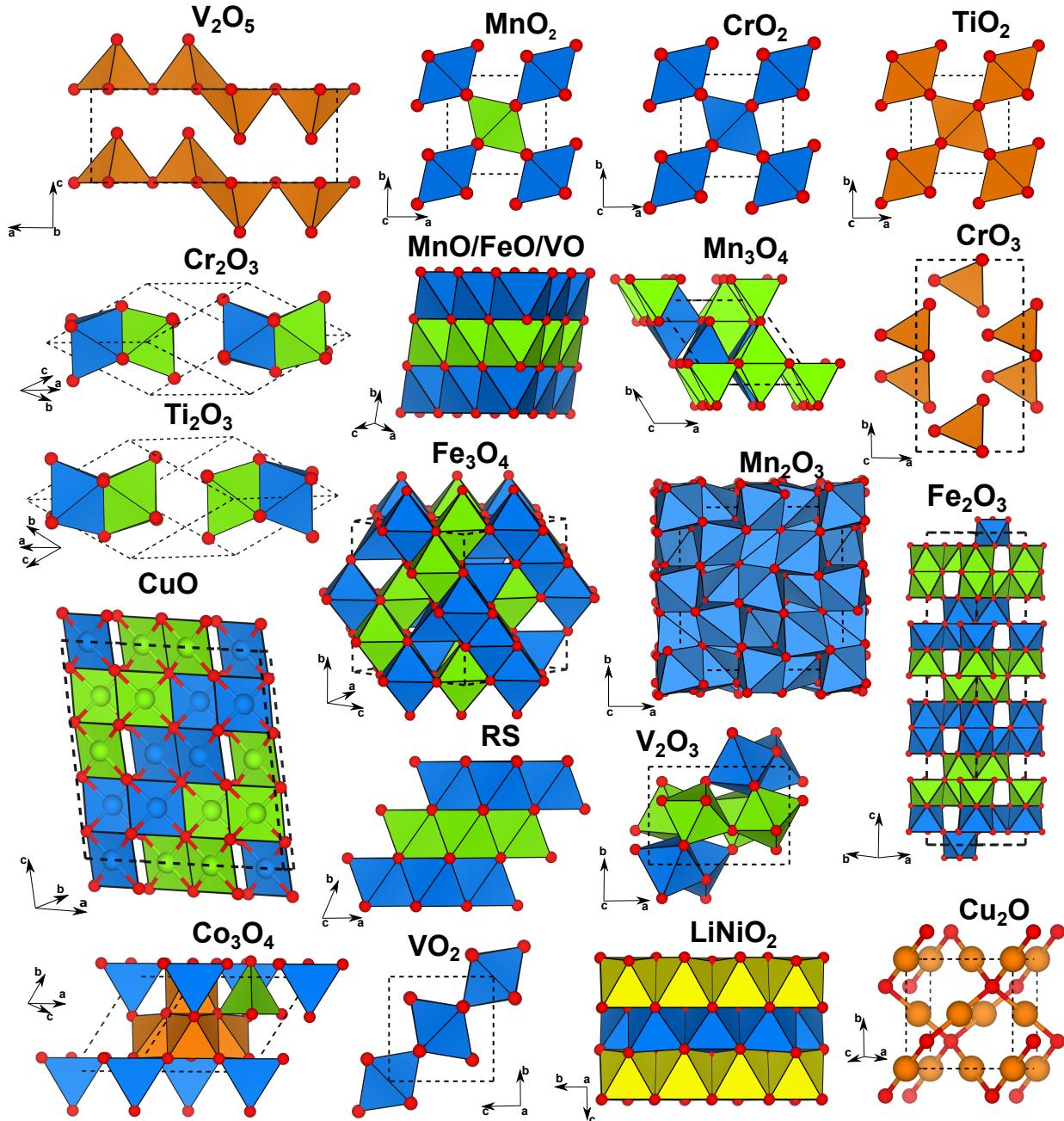


Figure S1: Crystal structures of TMOs considered in our work. Blue, green, and orange polyhedra represent TM centers with up, down, and no spin, respectively. Red spheres are oxygen atoms. Yellow polyhedra in LiNiO_2 represent Li atoms. RS refers to the $1 \times 2 \times 1$ supercell of the rocksalt structure of NiO and CoO .

Oxidation energetics of Cr and Cu oxides

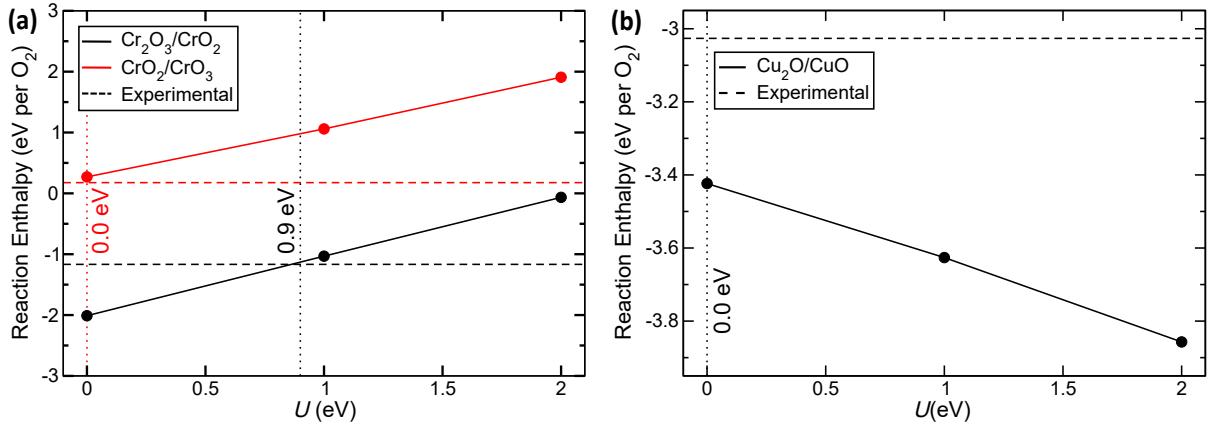


Figure S2: Oxidation enthalpy versus applied U within the restored regularized strongly constrained and appropriately normed (r^2 SCAN)+ U framework for (a) Cr and (b) Cu oxides. Notations on the plot are similar to those used in Figure 1 of the main text.

On-site magnetic moments

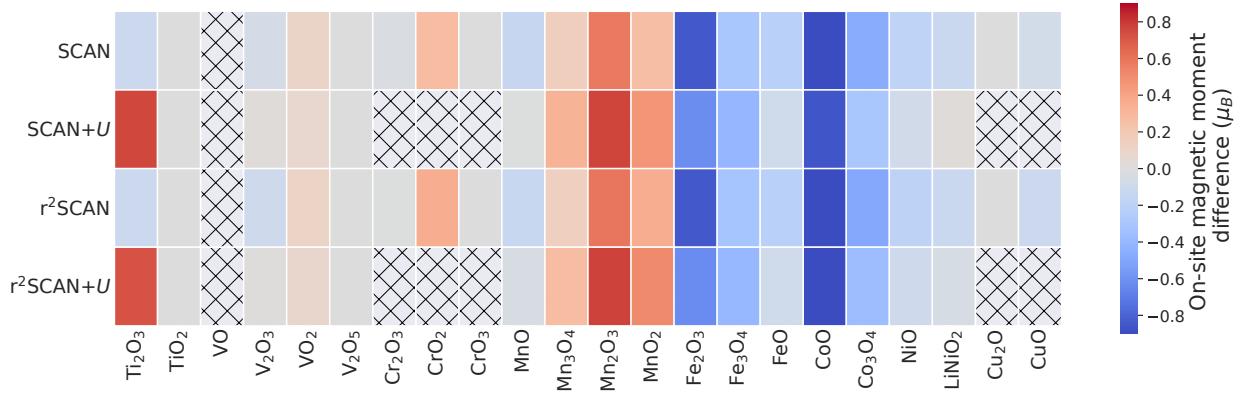


Figure S3: Difference between experimental and calculated on-site magnetic moments using the four functionals, plotted as a heatmap, for various systems. Red (blue) squares indicate overestimated (underestimated) calculated lattice volumes versus experiment.

Transferability checks

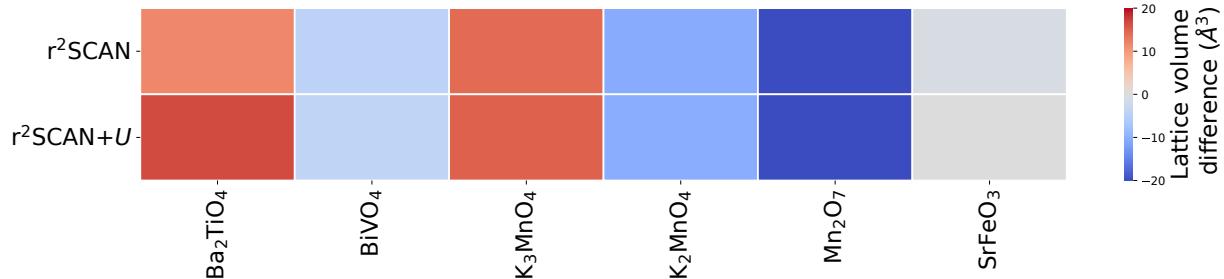


Figure S4: Difference between experimental and calculated lattice volumes (using $r^2\text{SCAN}$ and $r^2\text{SCAN}+U$), plotted as a heatmap, for various systems. Red (blue) squares indicate overestimated (underestimated) calculated lattice volumes versus experiment.

Table S1: Optimal U value for all the TMs obtained for the strongly constrained and appropriately normed (SCAN)+ U functional used from previous study.^{23,36}

Element	Optimal U (eV)
Ti	2.5
V	1.0
Cr	0.0
Mn	2.7
Fe	3.1
Co	3.0
Ni	2.5
Cu	0.0

Lattice parameters, on-site magnetic moments, and band gaps

Table S2: Lattice parameters, on-site magnetic moments, and band gaps obtained from experiments (denoted by ‘Expt.’) and calculated by SCAN, SCAN+ U , r²SCAN, and r²SCAN+ U for all TMOs considered in this study. The U values used with r²SCAN+ U and SCAN+ U are the corresponding optimal U values obtained for each TM, i.e., from Figure 1 of the main text and Table S1, respectively.

Composition (space group)	Source	Lattice constants (Å)			Lattice angles (°)			On-site moment (μ_B)	Band gap (eV)
		a	b	c	α	β	γ		
Ti_2O_3 $(R\bar{3}cR)$	Expt.	5.43	5.43	5.43	56.7	56.7	56.7	0.03-0.20	0.20
	SCAN	5.45	5.45	5.45	55.8	55.8	55.8	0.002	Metallic
	SCAN+ U	5.57	5.61	5.57	55.8	55.9	55.8	0.872	1.54
	r ² SCAN	5.46	5.46	5.46	55.7	55.7	55.7	0.002	Metallic
	r ² SCAN+ U	5.49	5.49	5.49	57.5	57.5	57.5	0.839	1.28
TiO_2 $(P4_2/mnm)$	Expt.	4.59	4.59	2.96	90.0	90.0	90.0	0	3.00
	SCAN	4.60	4.60	2.96	90.0	90.0	90.0	0	1.95
	SCAN+ U	4.61	4.61	3.00	90.0	90.0	90.0	0	2.21
	r ² SCAN	4.60	4.60	2.96	90.0	90.0	90.0	0	1.94
	r ² SCAN+ U	4.62	4.62	2.99	90.0	90.0	90.0	0	2.21
VO $(Fm\bar{3}m)$	Expt.	5.83	5.83	5.83	60.0	60.0	60.0	N/A	N/A
	SCAN	6.31	6.31	5.84	57.3	57.3	60.0	2.451	1.63
	SCAN+ U	6.31	6.31	5.94	57.9	57.9	60.0	2.564	2.12
	r ² SCAN	6.32	6.32	5.83	57.2	57.2	60.0	2.443	1.37
	r ² SCAN+ U	6.33	6.33	5.92	57.7	57.7	60.0	2.554	2.04

Continued on next page

Table S2: Lattice parameters, on-site magnetic moments, and band gaps obtained from experiments (denoted by ‘Expt.’) and calculated by SCAN, SCAN+ U , r²SCAN, and r²SCAN+ U for all TMOs considered in this study. The U values used with r²SCAN+ U and SCAN+ U are the corresponding optimal U values obtained for each TM, i.e., from Figure 1 of the main text and Table S1, respectively. (Continued)

	Expt.	7.26	5.00	5.55	90.0	96.8	90.0	1.20-2.37	0.20
V ₂ O ₃ (I2/a)								40,41	42
SCAN		7.23	5.03	5.56	90.0	96.8	90.0	1.728	Metallic
SCAN+ U		7.28	5.10	5.55	90.0	96.4	90.0	1.804	0.73
r ² SCAN		7.29	5.01	5.52	90.0	97.5	90.0	1.690	Metallic
r ² SCAN+ U		7.29	5.09	5.56	90.0	96.4	90.0	1.795	0.36
VO ₂ (P21/c)	Expt.	5.57	4.54	5.38	90.0	122.6	90.0	~1.00	0.70
SCAN		5.68	4.54	5.35	90.0	122.1	90.0	1.105	Metallic
SCAN+ U		5.83	4.52	5.35	90.0	122.3	90.0	1.074	Metallic
r ² SCAN		5.69	4.55	5.36	90.0	122.1	90.0	1.111	Metallic
r ² SCAN+ U		5.83	4.53	5.36	90.0	122.3	90.0	1.082	Metallic
V ₂ O ₅ (Pmmn)	Expt.	11.54	3.57	4.38	90.0	90.0	90.0	0	2.50
SCAN		11.56	3.55	4.28	90.0	90.0	90.0	0	1.69
SCAN+ U		11.57	3.56	4.28	90.0	90.0	90.0	0	1.80
r ² SCAN		11.60	3.55	4.32	90.0	90.0	90.0	0	1.72
r ² SCAN+ U		11.59	3.57	4.30	90.0	90.0	90.0	0	1.81
Cr ₂ O ₃ (R̄3cR)	Expt.	5.36	5.36	5.36	55.1	55.1	55.1	± 2.760	3.20
SCAN		5.36	5.36	5.36	55.0	55.0	55.0	2.736	1.89
r ² SCAN		5.38	5.38	5.38	54.8	54.8	54.8	2.755	1.71
CrO ₂	Expt.	4.42	4.42	2.92	90.0	90.0	90.0	2.00	Metallic

Continued on next page

Table S2: Lattice parameters, on-site magnetic moments, and band gaps obtained from experiments (denoted by ‘Expt.’) and calculated by SCAN, SCAN+ U , r²SCAN, and r²SCAN+ U for all TMOs considered in this study. The U values used with r²SCAN+ U and SCAN+ U are the corresponding optimal U values obtained for each TM, i.e., from Figure 1 of the main text and Table S1, respectively. (Continued)

								3	3
<i>(P4₂/mnm)</i>									
	SCAN	4.40	4.40	2.91	90.0	90.0	90.0	2.278	Metallic
	r ² SCAN	4.41	4.41	2.92	90.0	90.0	90.0	2.356	Metallic
CrO ₃	Expt.	4.79	8.56	5.74	90.0	90.0	90.0	0	3.80 ⁴⁷
<i>(C2cm)</i>									
	SCAN	4.68	8.52	5.70	90.0	90.0	90.0	0	1.89
	r ² SCAN	4.75	8.59	5.72	90.0	90.0	90.0	0	1.95
MnO	Expt.	6.29	6.29	6.29	60.0	60.0	60.0	4.580	3.60-3.80
<i>(Fm$\bar{3}m$)</i>								48	49
	SCAN	6.29	6.29	6.17	59.4	59.4	60.0	4.436	1.39
	SCAN+ U	6.31	6.31	6.24	59.6	59.6	60.0	4.576	1.97
	r ² SCAN	6.30	6.30	6.19	59.4	59.4	60.0	4.446	1.45
	r ² SCAN+ U	6.32	6.32	6.24	59.6	59.6	60.0	4.542	1.85
Mn ₃ O ₄	Expt.	5.75	6.22	5.75	117.5	90.0	117.5	4.34,3.25-3.64	2.30-2.50
<i>(I4₁/amd)</i>								50	51
	SCAN	5.75	6.19	5.75	117.4	90.5	117.4	3.638,4.383	0.96
	SCAN+ U	5.80	6.23	5.80	117.5	90.5	117.5	3.814,4.556	1.38
	r ² SCAN	5.76	6.21	5.76	117.4	90.5	117.4	3.683,4.375	0.80
	r ² SCAN+ U	5.80	6.24	5.80	117.5	90.4	117.5	3.807,4.504	1.06
Mn ₂ O ₃	Expt.	9.41	9.42	9.40	90.0	90.0	90.0	3.10-4.20	1.20-1.29
<i>(Pbca)</i>								31,52	53,54
	SCAN	9.40	9.42	9.41	90.0	90.0	90.0	3.681-3.799	Metallic
	SCAN+ U	9.45	9.48	9.47	90.0	90.0	90.0	3.864-3.923	0.24
	r ² SCAN	9.42	9.43	9.43	90.0	90.0	90.0	3.696-3.849	Metallic

Continued on next page

Table S2: Lattice parameters, on-site magnetic moments, and band gaps obtained from experiments (denoted by ‘Expt.’) and calculated by SCAN, SCAN+ U , r²SCAN, and r²SCAN+ U for all TMOs considered in this study. The U values used with r²SCAN+ U and SCAN+ U are the corresponding optimal U values obtained for each TM, i.e., from Figure 1 of the main text and Table S1, respectively. (Continued)

		r ² SCAN+ U	9.46	9.48	9.49	90.0	90.0	90.0	3.872-3.926	Metallic
MnO ₂	Expt.		4.40	4.40	2.87	90.0	90.0	90.0	2.350	0.27-0.3
(<i>P</i> 4 ₂ / <i>mnm</i>)								6		55,56
	SCAN		4.39	4.39	2.85	90.0	90.0	90.0	2.612	0.32
	SCAN+ U		4.40	4.40	2.88	90.0	90.0	90.0	2.821	0.66
	r ² SCAN		4.39	4.39	2.87	90.0	90.0	90.0	2.704	0.11
	r ² SCAN+ U		4.40	4.40	2.89	90.0	90.0	90.0	2.868	0.40
FeO	Expt.		6.08	6.08	6.08	60.0	60.0	60.0	3.32-4.20	2.40
(<i>Fm</i> 3 <i>m</i>)								57,58		59
	SCAN		5.85	5.97	6.25	61.4	60.4	59.3	3.542	0.17
	SCAN+ U		6.14	6.14	6.02	61.4	61.4	59.6	3.669	1.58
	r ² SCAN		5.97	5.88	6.13	61.4	62.4	60.6	3.542	0.14
	r ² SCAN+ U		6.14	6.13	6.13	60.3	59.6	60.1	3.675	1.47
Fe ₃ O ₄	Expt.		8.39	8.39	8.39	90.0	90.0	90.0	4.44,4.10	0.14
(<i>Fd</i> 3 <i>m</i>)								60		27
	SCAN		8.34	8.34	8.34	90.0	90.0	90.0	3.793,3.844	Metallic
	SCAN+ U		8.40	8.40	8.46	90.0	90.0	90.3	3.694,4.172	0.47
	r ² SCAN		8.36	8.36	8.36	90.2	89.5	89.4	3.782,3.839	Metallic
	r ² SCAN+ U		8.41	8.41	8.48	90.0	90.0	90.3	3.706,4.161	0.42
Fe ₂ O ₃	Expt.		5.04	5.04	27.54	90.0	90.0	120.0	4.90	2.20
(<i>R</i> 3 <i>c</i>)								61		62
	SCAN		5.03	5.03	27.49	90.0	90.0	120.0	4.056	0.60
	SCAN+ U		5.06	5.06	27.50	90.0	90.0	120.0	4.278	1.83

Continued on next page

Table S2: Lattice parameters, on-site magnetic moments, and band gaps obtained from experiments (denoted by ‘Expt.’) and calculated by SCAN, SCAN+ U , r²SCAN, and r²SCAN+ U for all TMOs considered in this study. The U values used with r²SCAN+ U and SCAN+ U are the corresponding optimal U values obtained for each TM, i.e., from Figure 1 of the main text and Table S1, respectively. (Continued)

	r ² SCAN	5.05	5.05	27.52	90.0	90.0	120.0	4.052	0.61
	r ² SCAN+ U	5.06	5.06	27.55	90.0	90.0	120.0	4.269	1.85
CoO	Expt.	3.01	6.01	3.01	60.0	60.0	60.0	± 3.35-3.80	2.40
(<i>Fm</i> $\bar{3}$ <i>m</i>)								58,63	64,65
	SCAN	3.00	5.99	3.00	59.9	60.0	60.0	2.581	0.91
	SCAN+ U	3.02	6.00	3.02	59.8	60.0	59.9	2.717	2.64
	r ² SCAN	2.99	5.99	2.99	60.0	60.0	60.0	2.575	0.61
	r ² SCAN+ U	3.01	6.00	3.01	59.9	60.0	59.9	2.668	1.98
Co ₃ O ₄	Expt.	5.71	5.71	5.71	60.0	60.0	60.0	± 3.02,0.00	1.60
(<i>Fd</i> $\bar{3}$ <i>m</i>)								30	64
	SCAN	5.69	5.69	5.69	60.0	60.0	60.0	2.549	1.20
	SCAN+ U	5.71	5.71	5.71	60.0	60.0	60.0	2.716	1.23
	r ² SCAN	5.69	5.69	5.69	60.0	60.0	60.0	2.530	0.81
	r ² SCAN+ U	5.71	5.71	5.71	60.0	60.0	60.0	2.653	0.83
NiO	Expt.	2.95	5.91	2.95	60.0	60.0	60.0	± 1.64-1.90	4.30
(<i>Fm</i> $\bar{3}$ <i>m</i>)								48,66	67
	SCAN	2.94	5.87	2.94	60.0	60.0	60.0	1.585	2.39
	SCAN+ U	2.95	5.89	2.95	60.0	60.0	60.0	1.690	3.35
	r ² SCAN	2.94	5.87	2.94	60.0	60.0	60.0	1.577	2.11
	r ² SCAN+ U	2.94	5.89	2.94	60.0	60.0	60.0	1.677	3.17
LiNiO ₂	Expt.	5.01	5.01	5.02	80.4	70.6	60.0	1.00	N/A
(<i>P1</i> m <i>1</i>)								68,69	
	SCAN	5.11	4.87	5.06	78.5	67.7	58.3	0.879	Metallic

Continued on next page

Table S2: Lattice parameters, on-site magnetic moments, and band gaps obtained from experiments (denoted by ‘Expt.’) and calculated by SCAN, SCAN+ U , r²SCAN, and r²SCAN+ U for all TMOs considered in this study. The U values used with r²SCAN+ U and SCAN+ U are the corresponding optimal U values obtained for each TM, i.e., from Figure 1 of the main text and Table S1, respectively. (Continued)

	SCAN+ U	5.10	4.87	5.07	78.6	67.3	58.4	1.020	0.13
	r ² SCAN	4.97	4.94	4.99	80.2	70.2	59.8	0.877	Metallic
	r ² SCAN+ U	5.10	4.87	5.07	78.6	67.8	58.4	0.956	0.08
Cu ₂ O	Expt.	4.27	4.27	4.27	90.0	90.0	90.0	0	2.17-2.4
(<i>Pn</i> 3̄ <i>m</i>)									34,70
	SCAN	4.23	4.23	4.23	90.0	90.0	90.0	0	0.60
	r ² SCAN	4.23	4.23	4.23	90.0	90.0	90.0	0	0.58
CuO	Expt.	9.37	3.42	10.26	90.0	99.5	90.0	0.680	1.40
(<i>C</i> 2/ <i>c</i>)								71	34
	SCAN	8.80	3.70	10.26	90.0	96.3	90.0	0.607	0.79
	r ² SCAN	8.67	3.79	10.27	90.0	96.1	90.0	0.566	0.46

Table S3: Lattice parameters reported experimentally (denoted by ‘Expt.’) and calculated by r²SCAN, and r²SCAN+*U* for all TMOs considered as transferability checks. The *U* values used with r²SCAN+*U* are the corresponding optimal *U* values obtained for each TM (from Figure 1 of the main text).

Composition (space group)	Source	Lattice constants (Å)			Lattice angles (°)		
		a	b	c	α	β	γ
Ba_2TiO_4 $(P2_1/n)$	Expt.	6.09	7.68	10.54	90.0	93.0	90.0
	r ² SCAN	6.17	7.73	10.60	90.0	93.6	90.0
	r ² SCAN+ <i>U</i>	6.20	7.75	10.63	90.0	93.9	90.0
BiVO_4 $(I2/b)$	Expt.	5.19	5.09	11.69	90.0	90.0	90.4
	r ² SCAN	5.13	5.13	11.59	90.0	90.0	90.0
	r ² SCAN+ <i>U</i>	5.13	5.13	11.58	90.0	90.0	90.0
K_3MnO_4 $(I\bar{4}2m)$	Expt.	5.90	5.90	8.01	90.0	90.0	90.0
	r ² SCAN	6.51	6.50	6.92	90.0	90.0	90.0
	r ² SCAN+ <i>U</i>	6.50	6.50	6.94	90.0	90.0	90.0
K_2MnO_4 $(Pnma)$	Expt.	7.67	5.90	10.36	90.0	90.0	90.0
	r ² SCAN	7.64	5.81	10.32	90.0	90.0	90.0
	r ² SCAN+ <i>U</i>	7.64	5.81	10.32	90.0	90.0	90.0
Mn_2O_7 $(P2_1/c)$	Expt.	6.80	16.69	9.45	90.0	100.2	90.0
	r ² SCAN	6.72	16.68	9.20	90.0	99.5	90.0
	r ² SCAN+ <i>U</i>	6.73	16.66	9.19	90.0	99.5	90.0
SrFeO_3 $(Pm\bar{3}m)$	Expt.	3.85	3.85	3.85	90.0	90.0	90.0
	r ² SCAN	3.83	3.83	3.83	90.0	90.0	90.0
	r ² SCAN+ <i>U</i>	3.85	3.85	3.85	90.0	90.0	90.0

Computational time

Computational time required per ionic step for a TMO is calculated by dividing the overall computational time by the total number of ionic steps completed during its structural relaxation. Similarly, computational time required per electronic step for a TMO is calculated by dividing the overall computational time by the total number of electronic steps completed during its structural relaxation. Relative computational time is subsequently calculated as the computational time (overall, per ionic step, or per electronic step) of a given functional divided by the corresponding computational time (overall, per ionic step, or per electronic step) taken by the SCAN functional. Thus a value of 1.0 on the computational time scale indicates that the functional takes an identical amount of time for structure relaxation as SCAN, while values lower (greater) than 1.0 indicate lower (greater) computational time than SCAN. For a given TM-O₂ binary system, we simply averaged the relative computational times across the individual TMOs constituting the binary system. All computational times were normalized per number of computing cores used in a calculation.

Density of states

The electronic density of states (DOS) for all TMOs not illustrated in Figure 2 of the main text are included in this section, from Figures S5 to S21. In each DOS plot, solid orange lines correspond to the $2p$ -states of O, while solid green lines indicate the $3d$ -states of the TM. Li s states are not displayed in Figure S20. The dashed black line represents the Fermi level. The dotted vertical lines represent the valence and conduction band edges. Magnitude of the band gap in all gapped systems is indicated by the text annotation next to the conduction band minimum. The zero on the energy scale is set to the valence band maximum for TMOs with a finite band gap and to the Fermi level in metallic systems.

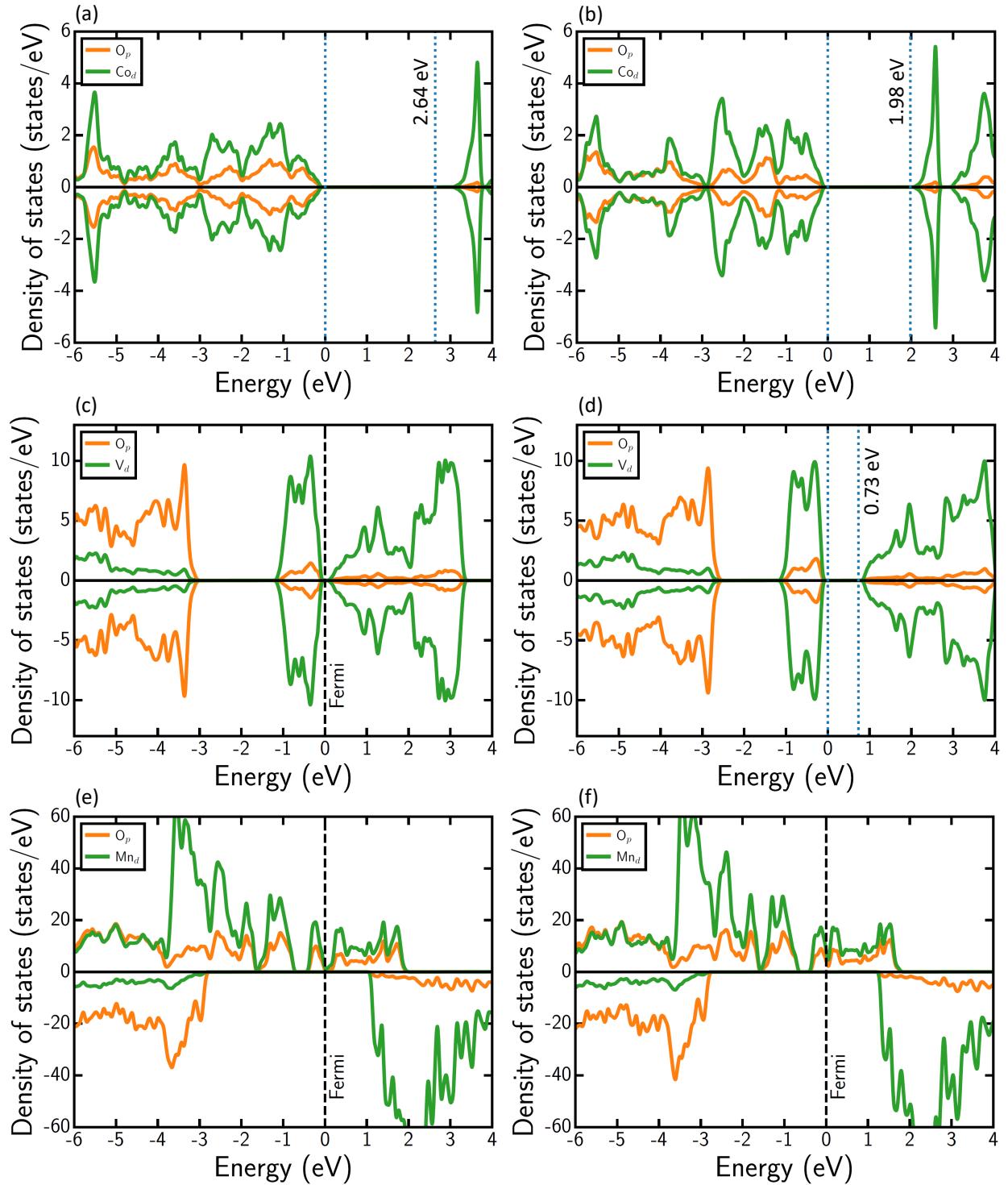


Figure S5: DOS plots for CoO calculated using (a) SCAN+ U , and (b) r^2 SCAN+ U . DOS plots for V_2O_3 calculated using (c) SCAN, and (d) SCAN+ U . DOS plots for Mn_2O_3 calculated using (e) SCAN and (f) r^2 SCAN.

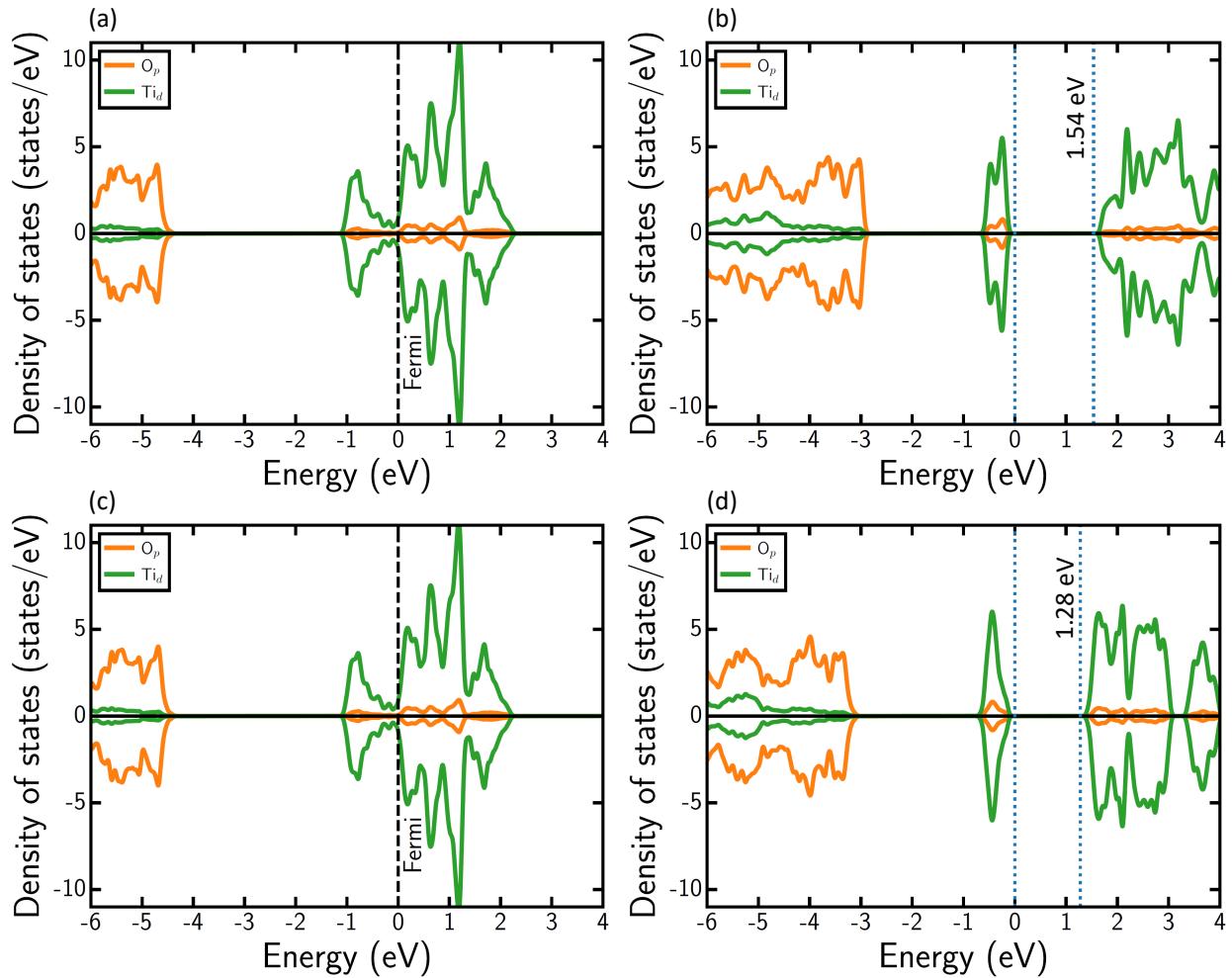


Figure S6: DOS plots for Ti_2O_3 calculated using (a) SCAN, (b) SCAN+ U , (c) $r^2\text{SCAN}$ and (d) $r^2\text{SCAN}+U$.

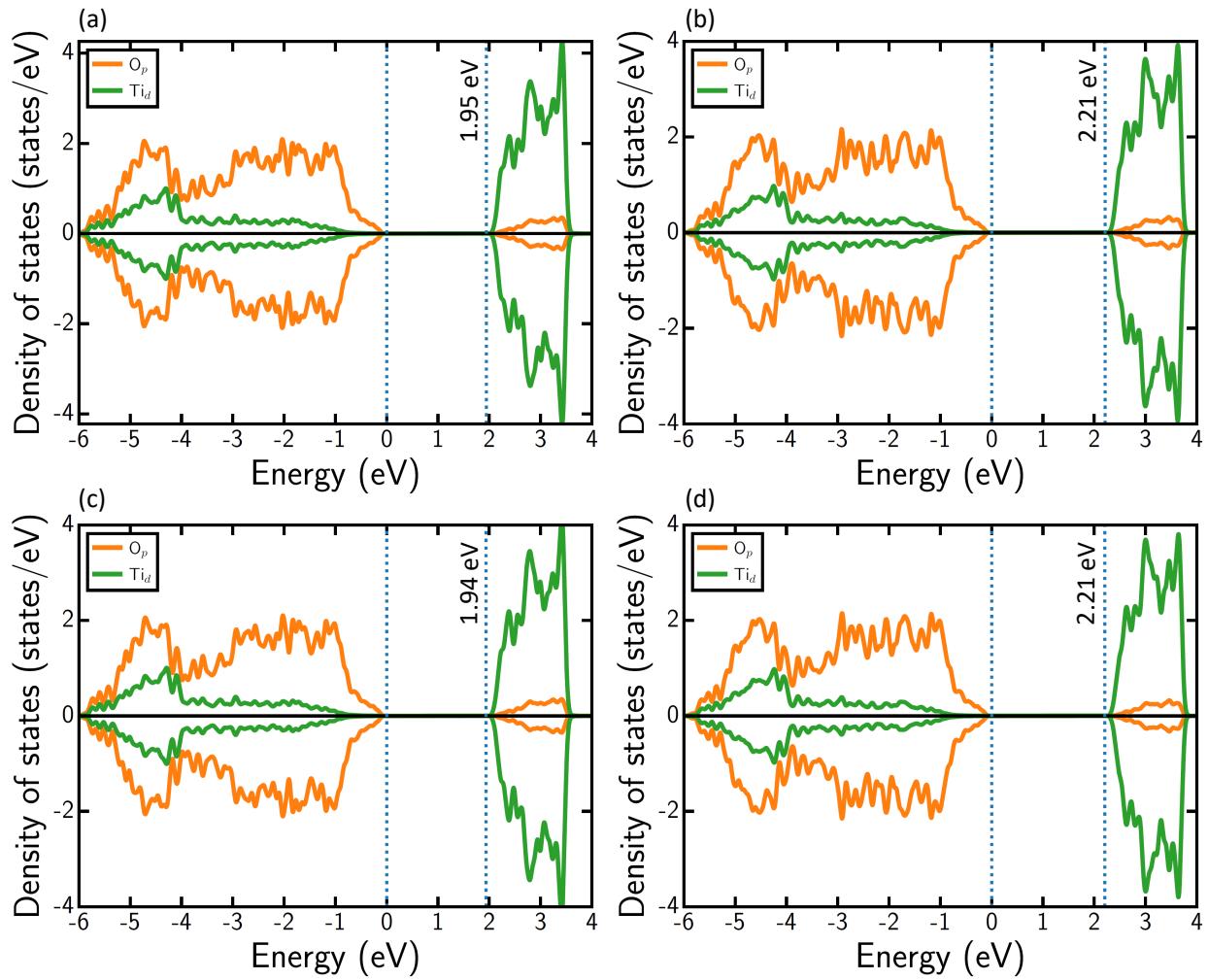


Figure S7: DOS plots for TiO_2 calculated using (a) SCAN, (b) SCAN+ U , (c) $r^2\text{SCAN}$ and (d) $r^2\text{SCAN}+U$.

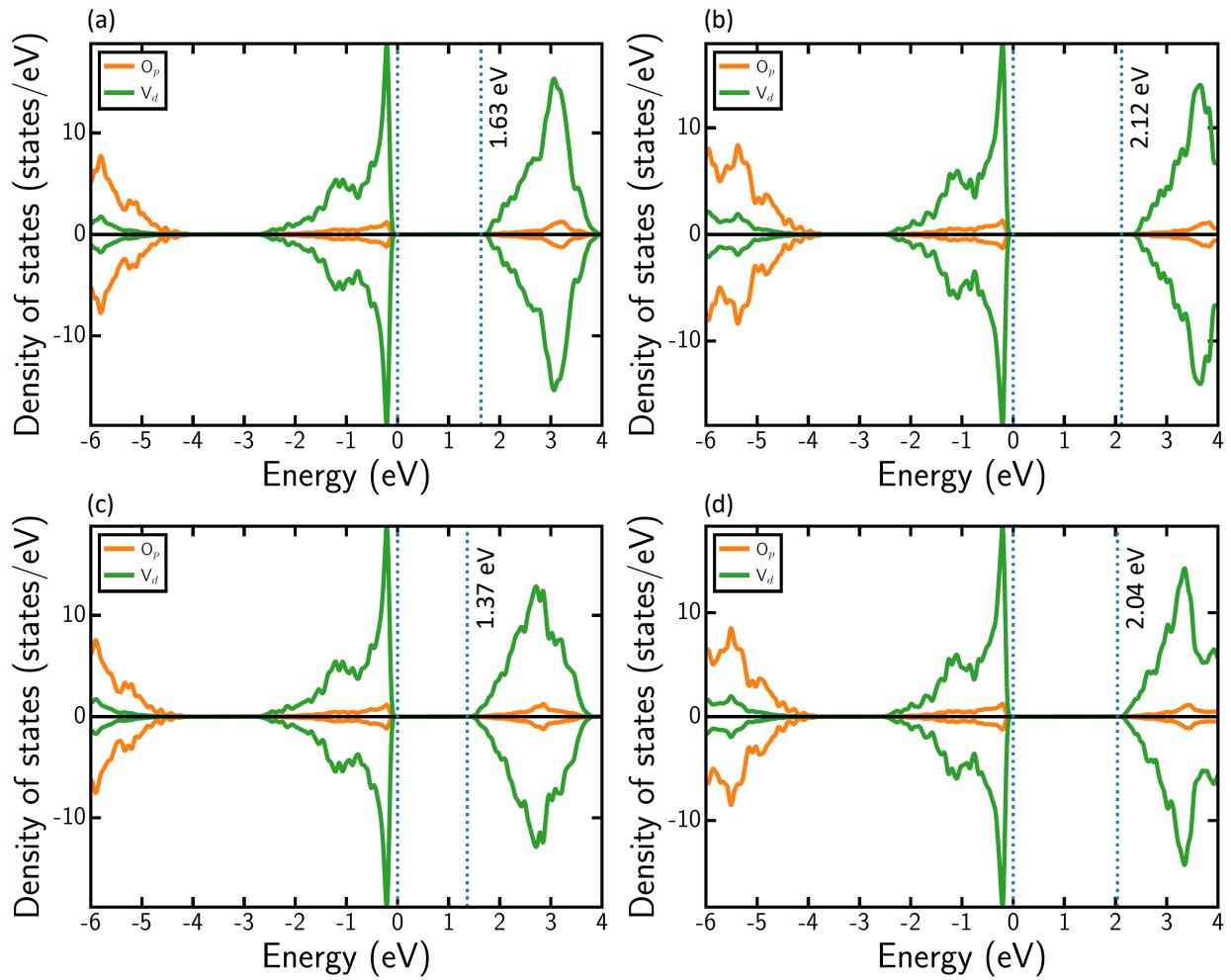


Figure S8: DOS plots for VO calculated using (a) SCAN, (b) SCAN+ U , (c) r^2 SCAN and (d) r^2 SCAN+ U .

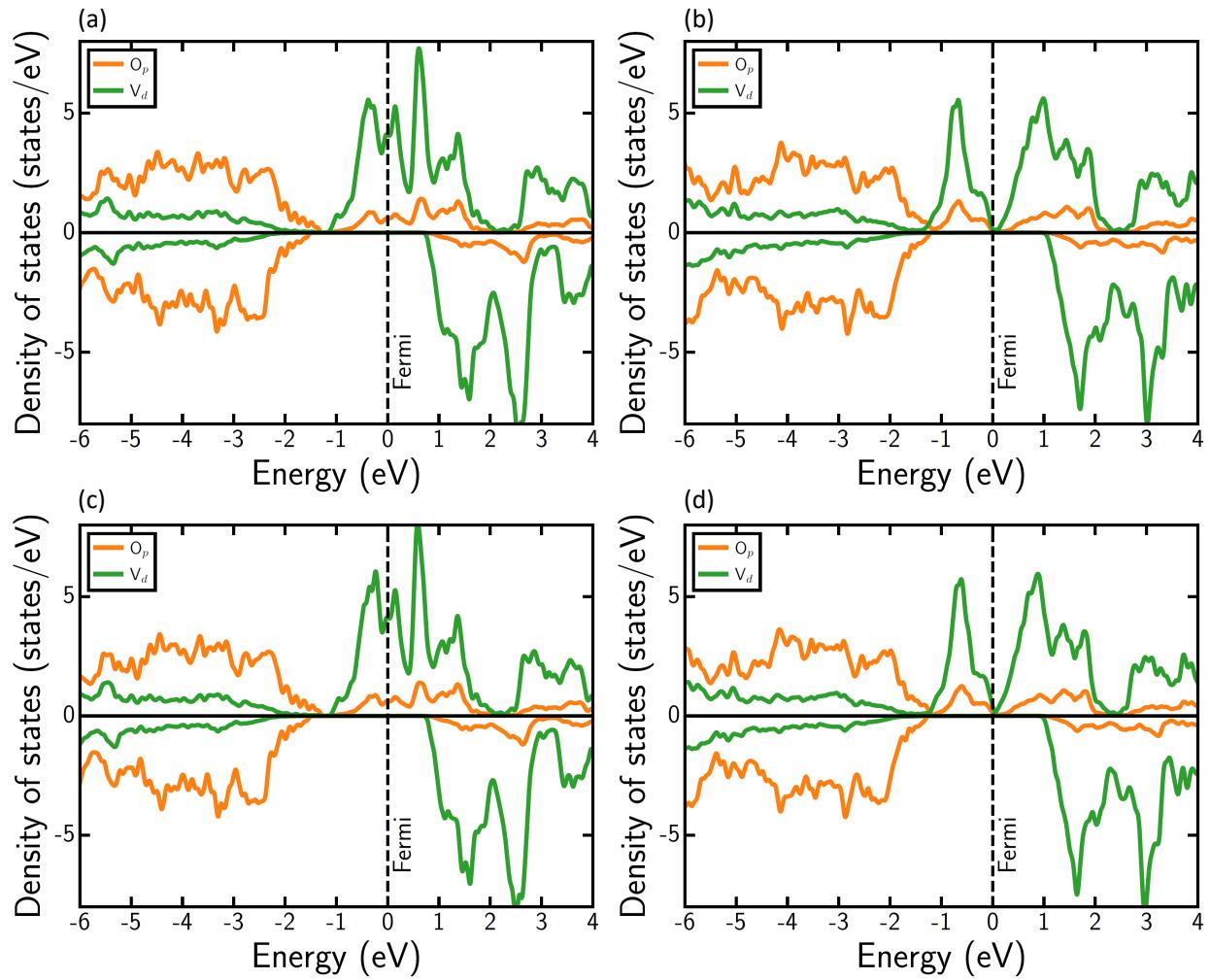


Figure S9: DOS plots for VO_2 calculated using (a) SCAN, (b) SCAN+ U , (c) $r^2\text{SCAN}$ and (d) $r^2\text{SCAN}+U$.

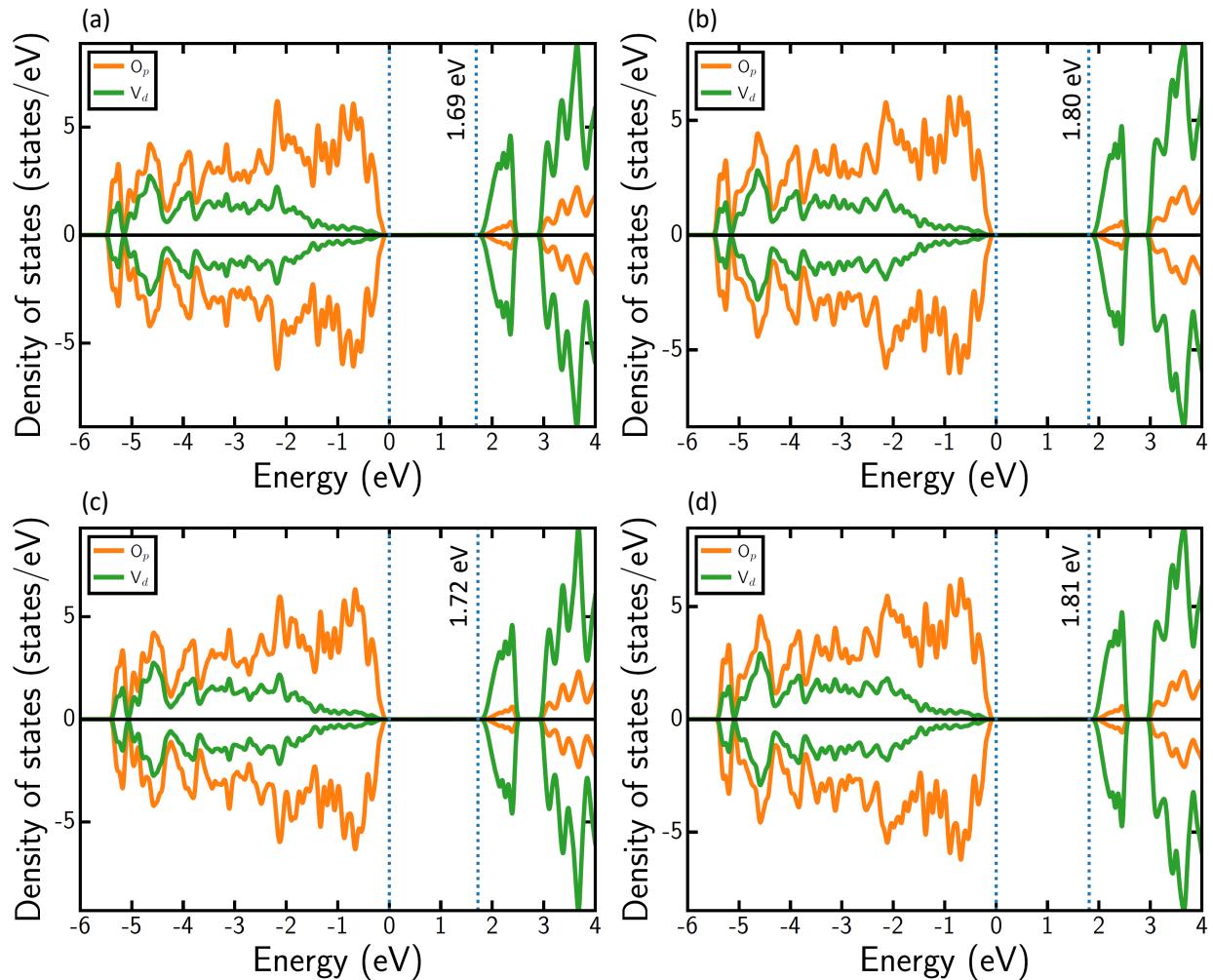


Figure S10: DOS plots for V_2O_5 calculated using (a) SCAN, (b) SCAN+ U , (c) r^2 SCAN and (d) r^2 SCAN+ U .

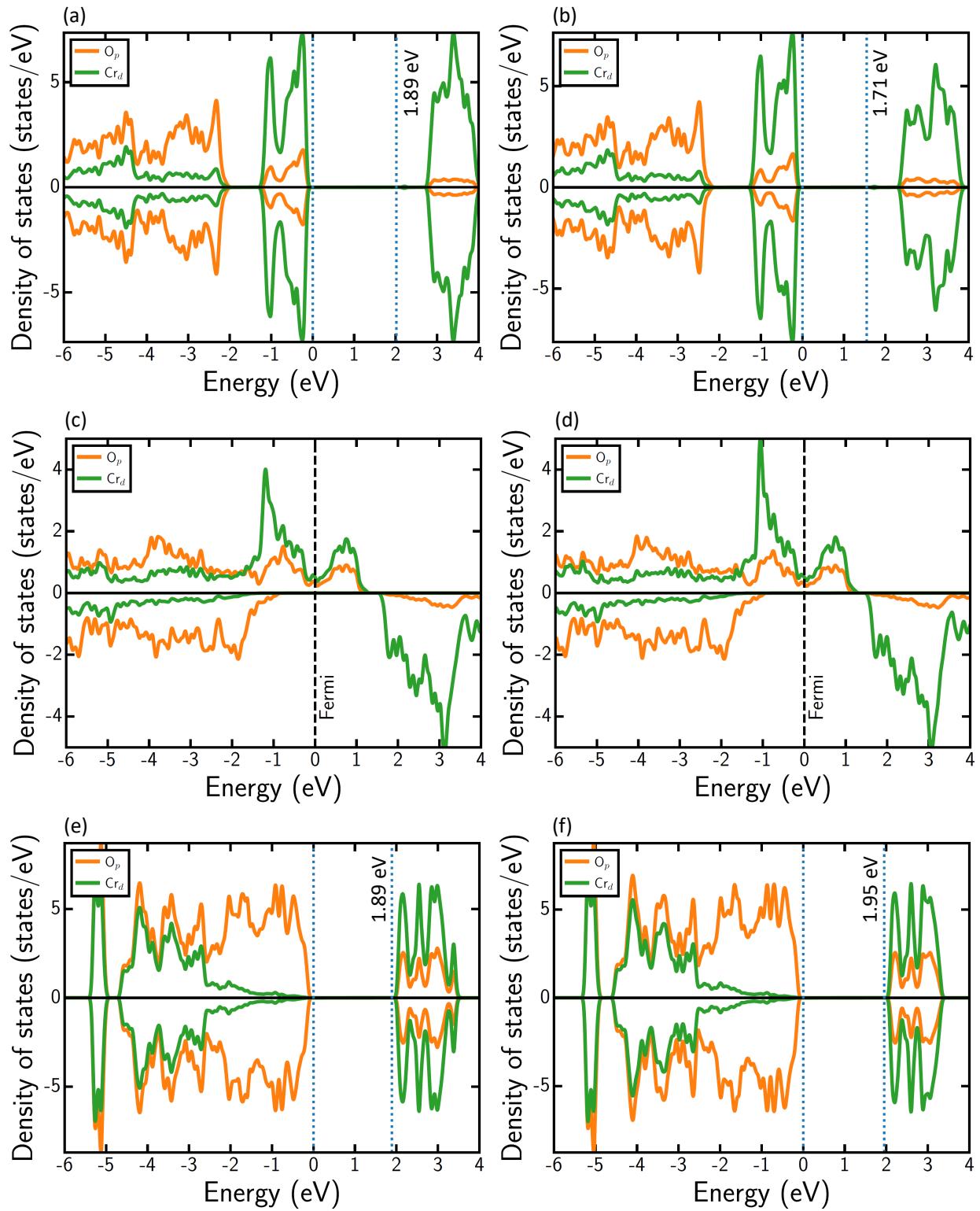


Figure S11: DOS plots for Cr_2O_3 (panels a and b), CrO_2 (c and d), and CrO_3 (e and f), calculated using SCAN (panels a, c, and e) and $r^2\text{SCAN}$ (b, d, and f).

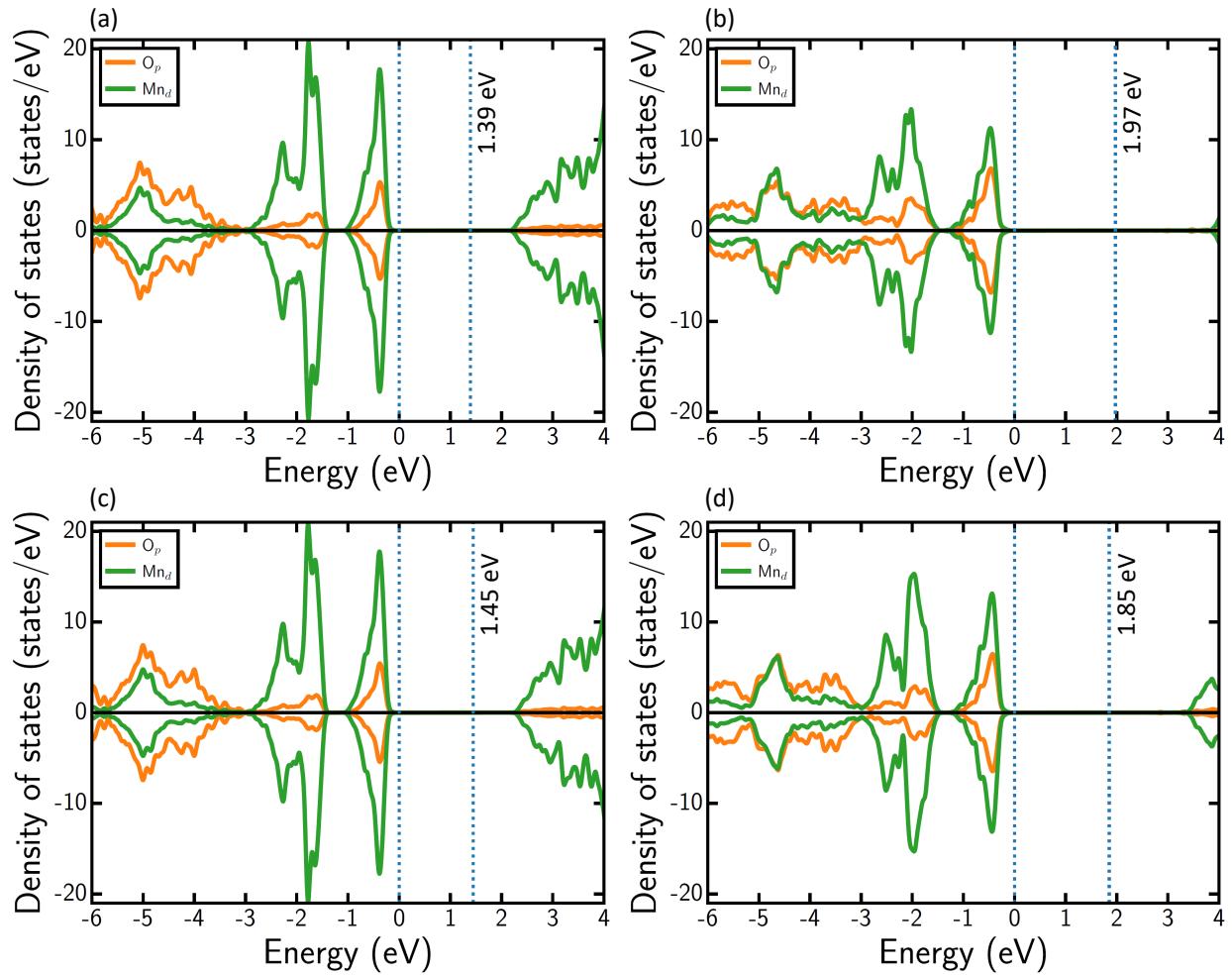


Figure S12: DOS plots for MnO calculated using (a) SCAN, (b) SCAN+ U , (c) r^2 SCAN and (d) r^2 SCAN+ U .

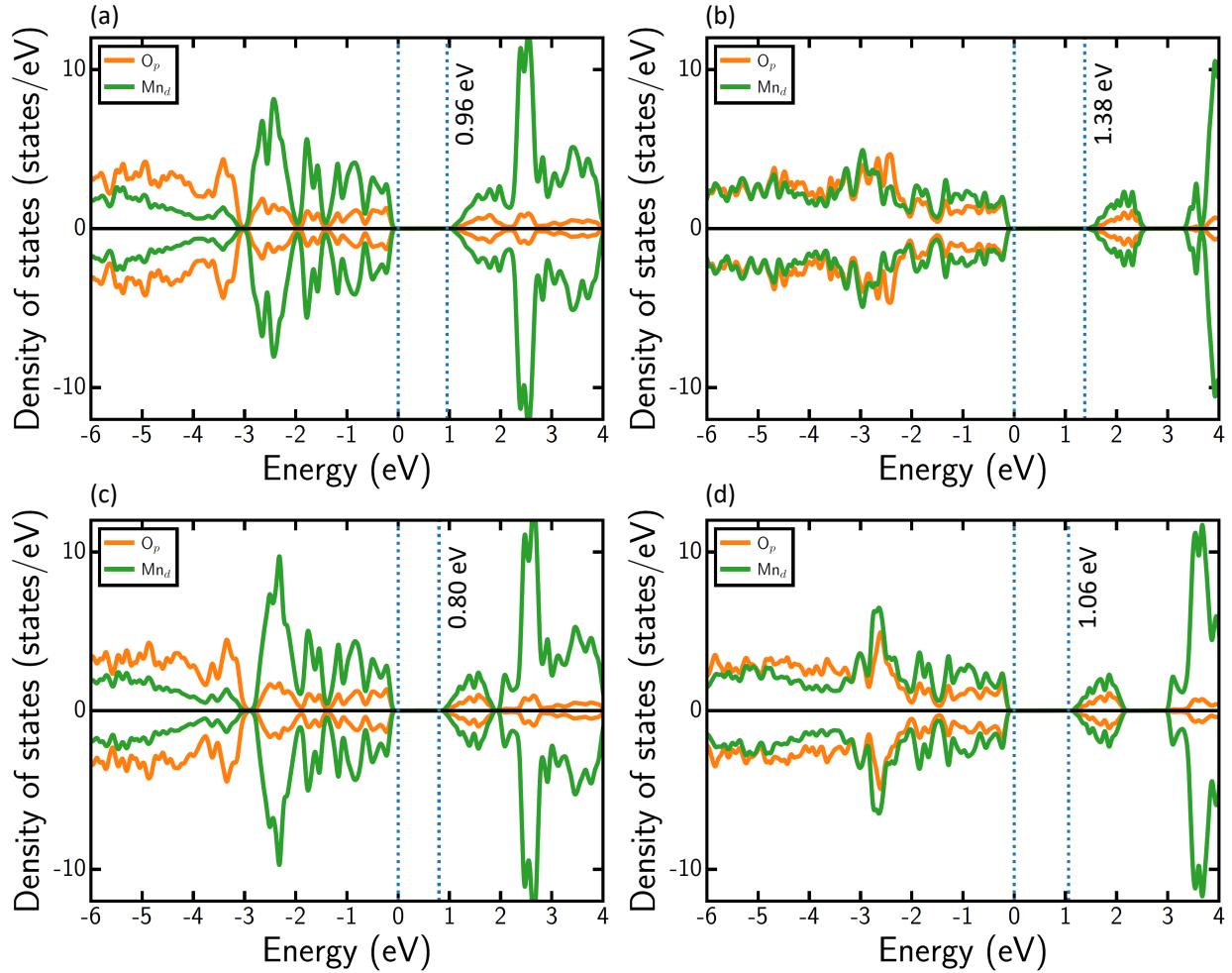


Figure S13: DOS plots for Mn_3O_4 calculated using (a) SCAN, (b) SCAN+ U , (c) r^2 SCAN and (d) r^2 SCAN+ U .

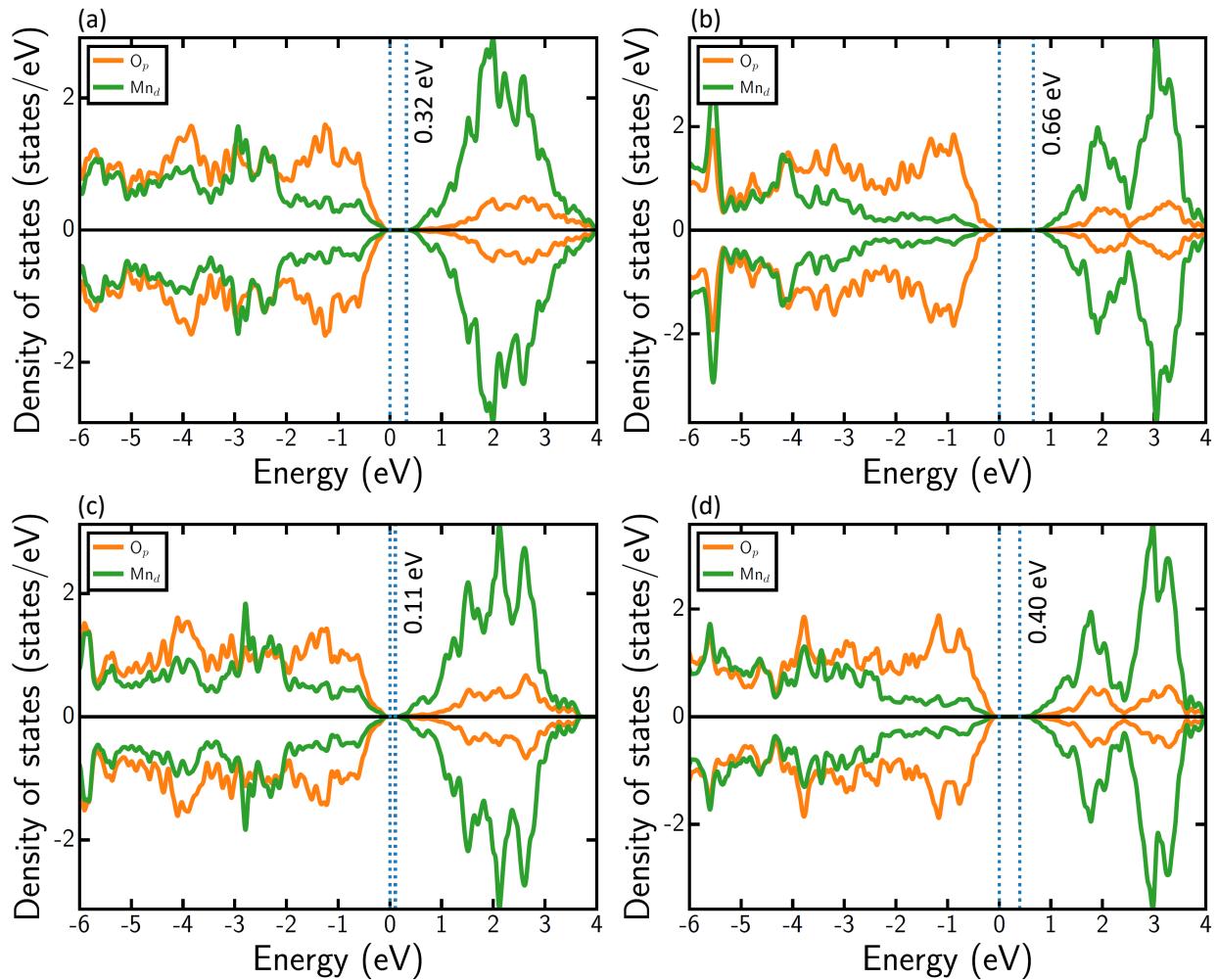


Figure S14: DOS plots for MnO_2 calculated using (a) SCAN, (b) SCAN+ U , (c) $r^2\text{SCAN}$ and (d) $r^2\text{SCAN}+U$.

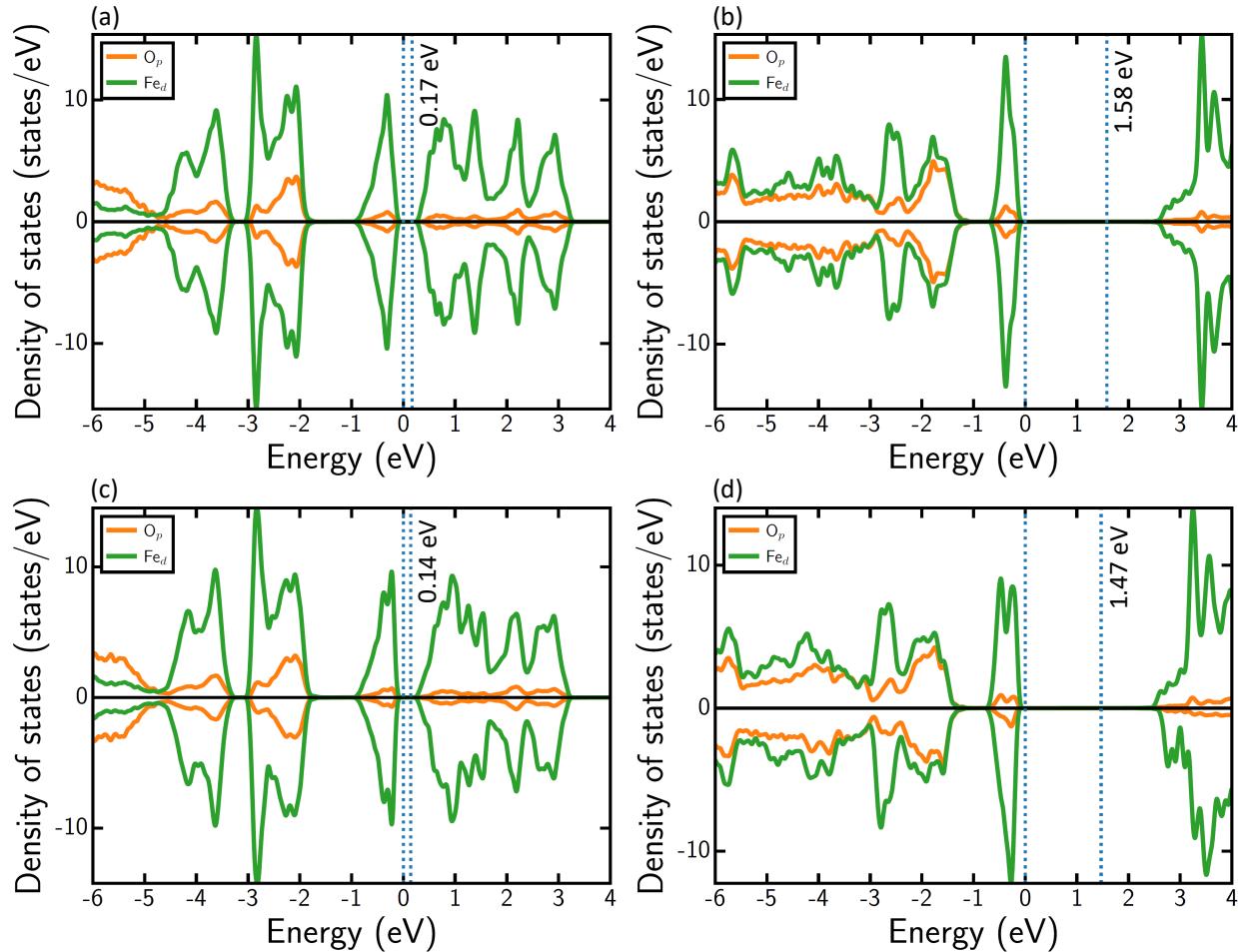


Figure S15: DOS plots for FeO calculated using (a) SCAN, (b) SCAN+ U , (c) r^2 SCAN and (d) r^2 SCAN+ U .

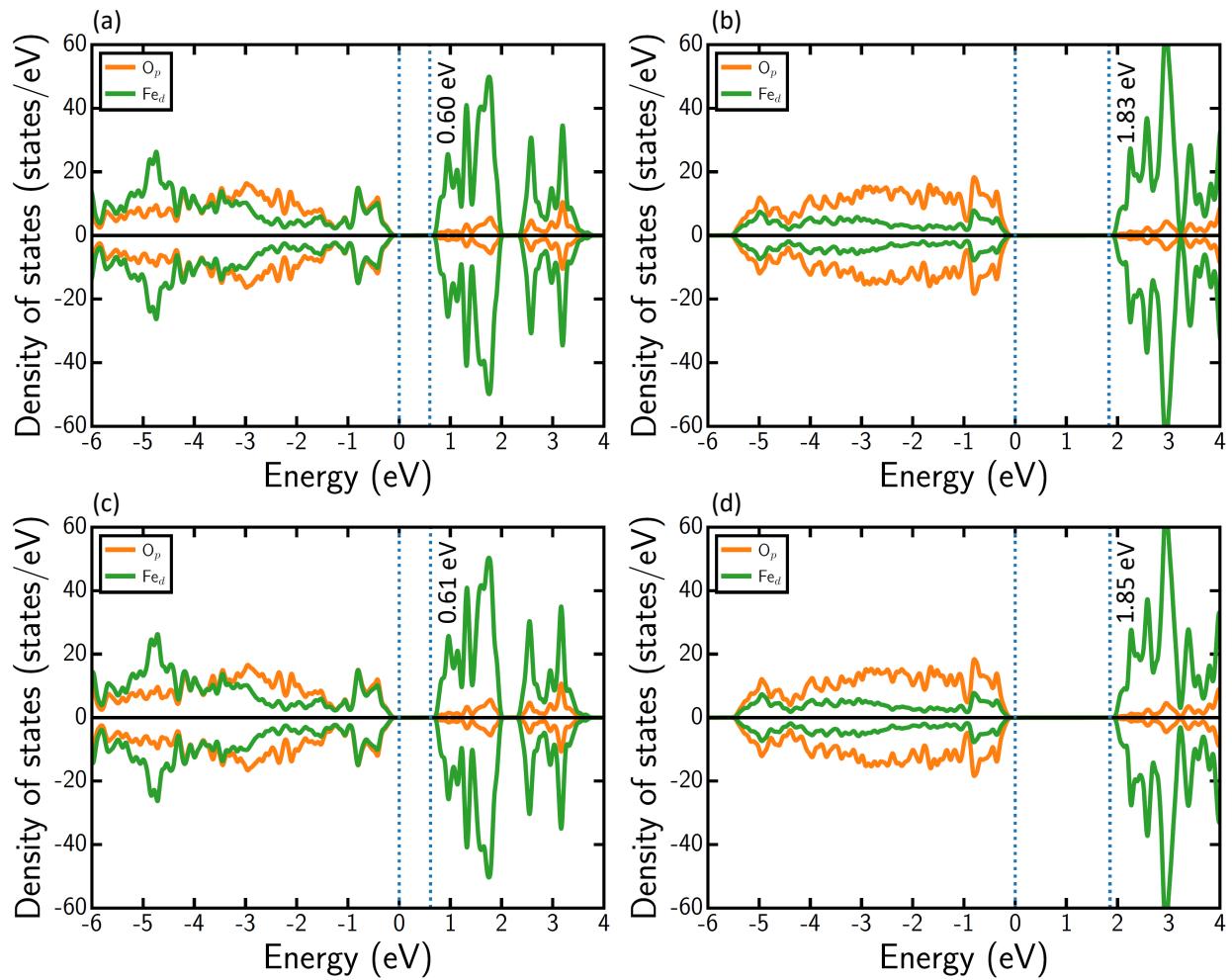


Figure S16: DOS plots for Fe_2O_3 calculated using (a) SCAN, (b) SCAN+ U , (c) r^2 SCAN and (d) r^2 SCAN+ U .

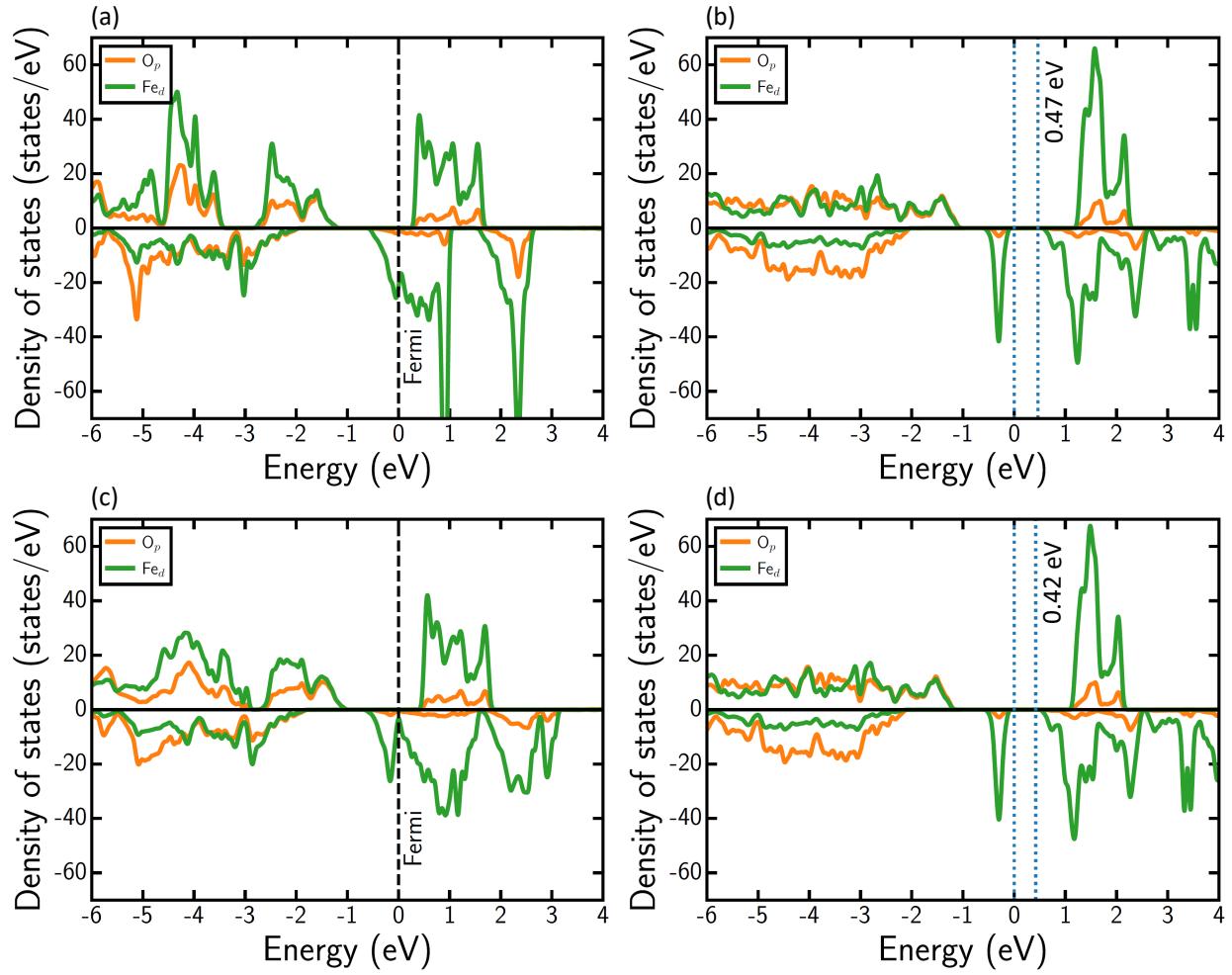


Figure S17: DOS plots for Fe_3O_4 calculated using (a) SCAN, (b) SCAN+ U , (c) r^2 SCAN and (d) r^2 SCAN+ U .

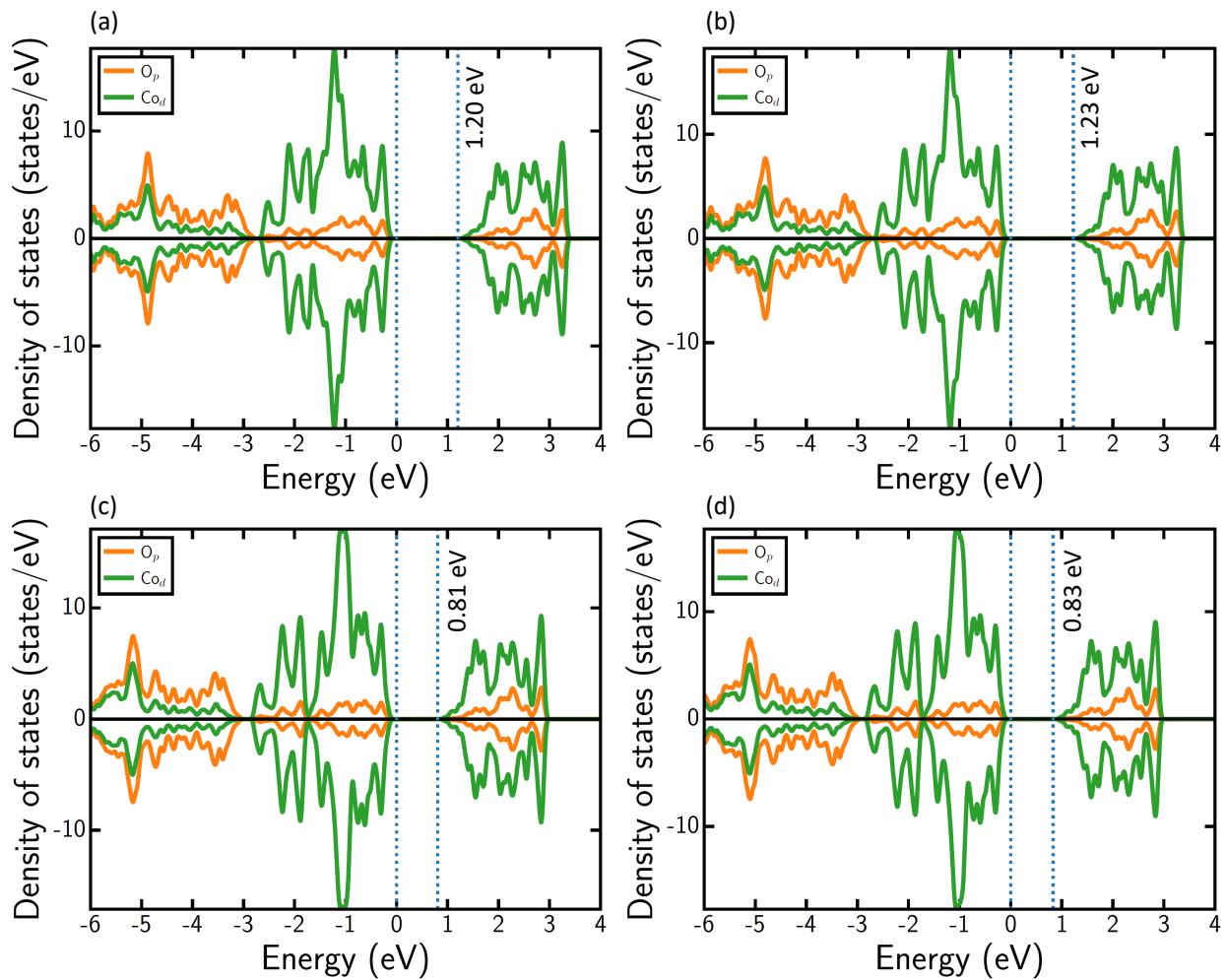


Figure S18: DOS plots for Co_3O_4 calculated using (a) SCAN, (b) SCAN+ U , (c) $r^2\text{SCAN}$ and (d) $r^2\text{SCAN}+U$.

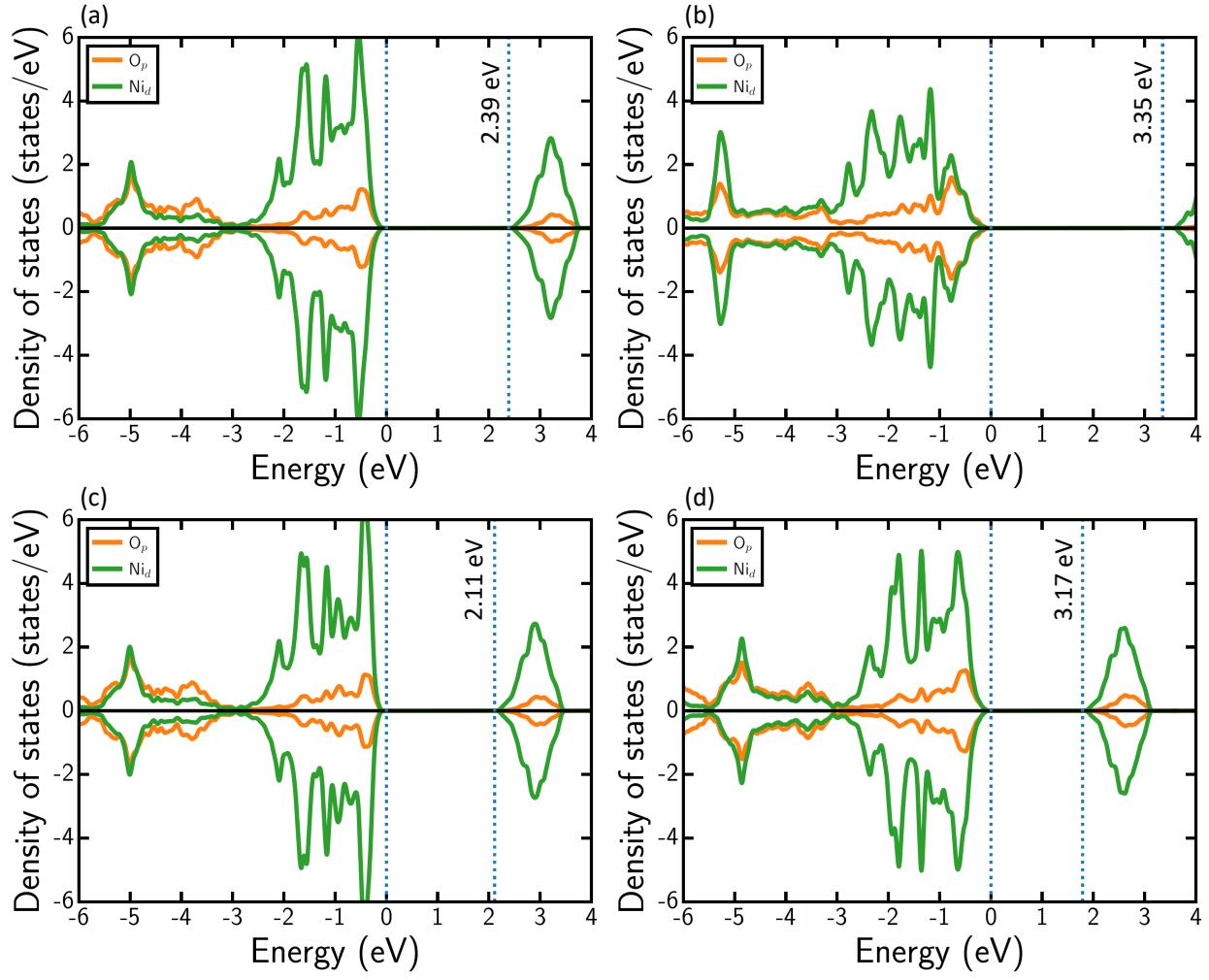


Figure S19: DOS plots for NiO calculated using (a) SCAN, (b) SCAN+ U , (c) r^2 SCAN and (d) r^2 SCAN+ U .

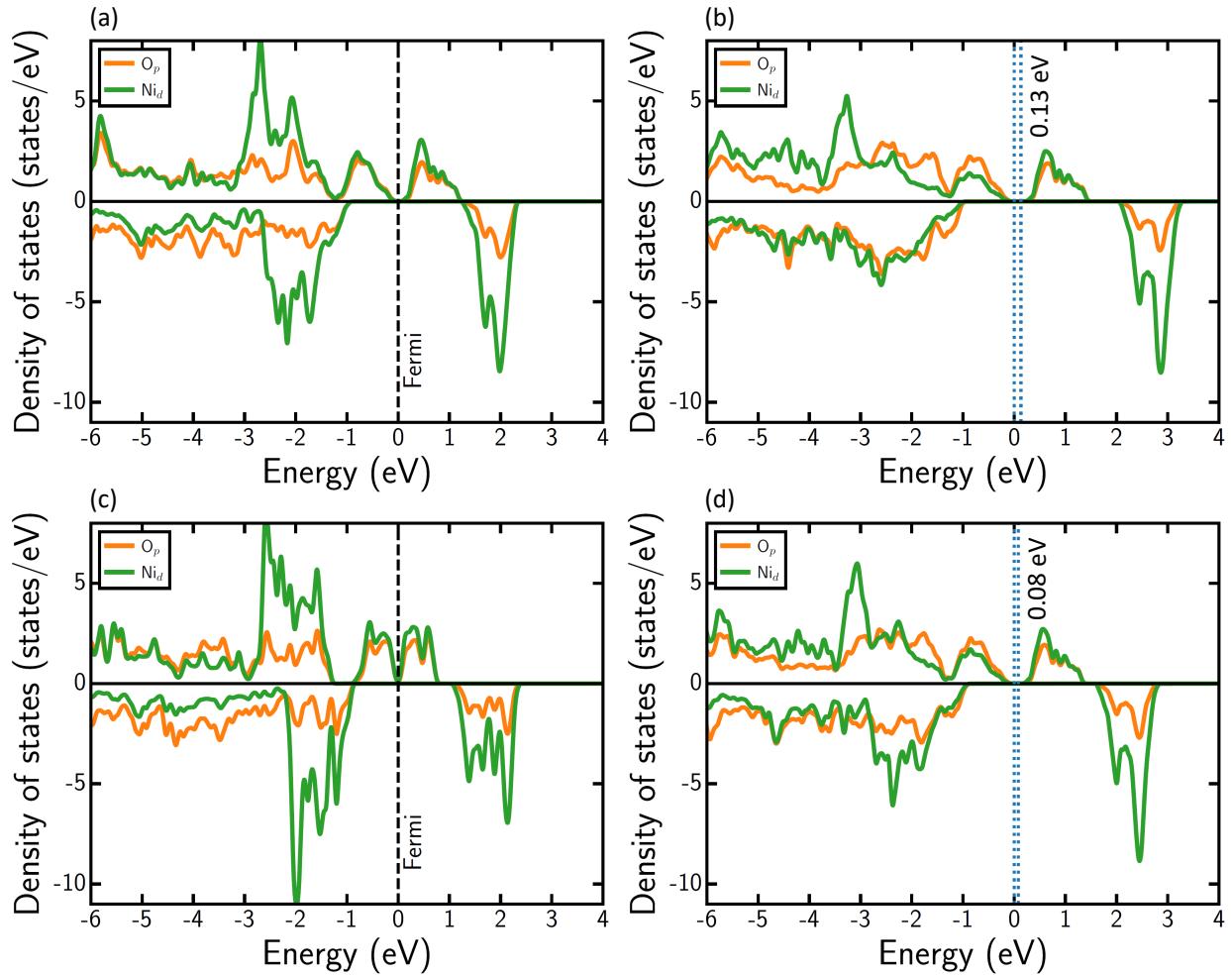


Figure S20: DOS plots for LiNiO_2 calculated using (a) SCAN, (b) SCAN+ U , (c) r^2 SCAN and (d) r^2 SCAN+ U .

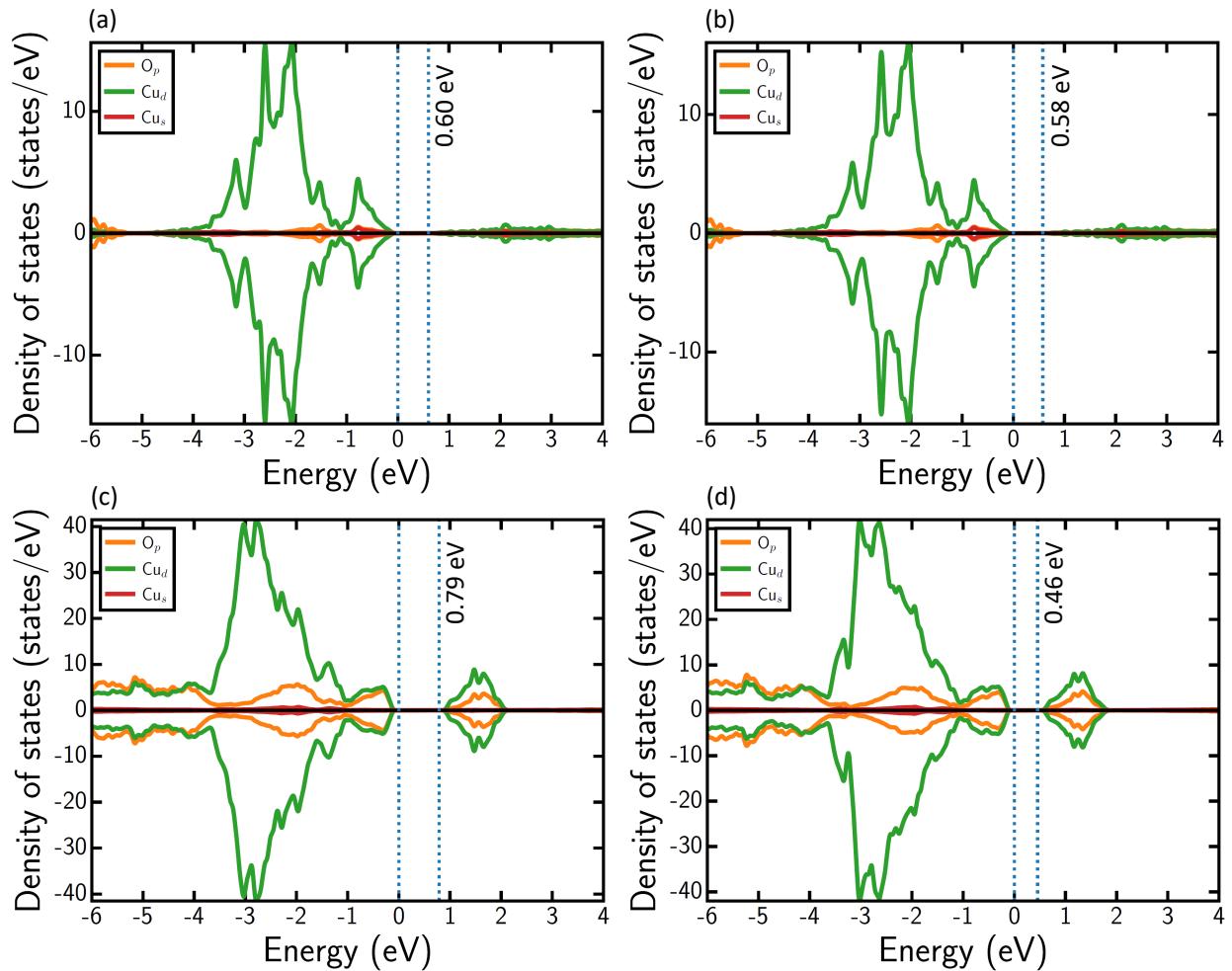


Figure S21: DOS plots for Cu_2O (panels a and b), and CuO (c and d), calculated using SCAN (panels a and c) and $r^2\text{SCAN}$ (b and d).

References

- (1) Baur, W. H. Rutile-Type Compounds. V. Refinement of MnO₂ and MgF₂. *Acta Crystallogr., Sect. B* **1976**, *32*, 2200.
- (2) Kang, W.; Hybertsen, M. S. Quasiparticle and optical properties of rutile and anatase TiO₂. *Phys. Rev. B* **2010**, *82*, 085203.
- (3) Coey, J.; Venkatesan, M. Half-metallic ferromagnetism: Example of CrO₂. *J. Appl. Phys.* **2002**, *91*, 8345–8350.
- (4) Voinov, M. MnO₂: structure and activity. *Electrochim. Acta* **1982**, *27*, 833–835.
- (5) Porta, P.; Marezio, M.; Remeika, J.; Dernier, P. D. Chromium dioxide: High pressure synthesis and bond lengths. *Materials Research Bulletin* **1972**, *7*, 157–161.
- (6) Regulski, M.; Przeniosło, R.; Sosnowska, I.; Hoffmann, J.-U. Incommensurate magnetic structure of β -MnO₂. *Phys. Rev. B* **2003**, *68*, 172401.
- (7) Rogers, K. D. An X-ray diffraction study of semiconductor and metallic vanadium dioxide. *Powder Diffr.* **1993**, *8*, 240–244.
- (8) Robinson, W. R. The crystal structures of Ti₂O₃, a semiconductor, and (Ti_{0.900}V_{0.100})₂O₃, a semimetal. *J. Solid State Chem.* **1974**, *9*, 255–260.
- (9) Crawford, J. A.; Vest, R. W. Electrical conductivity of single-crystal Cr₂O₃. *Journal of Applied Physics* **1964**, *35*, 2413–2418.
- (10) Brockhouse, B. N. Antiferromagnetic structure in Cr₂O₃. *J. Chem. Phys.* **1953**, *21*, 961–962.
- (11) Abrahams, S. C. Magnetic and crystal structure of titanium sesquioxide. *Phys. Rev.* **1963**, *130*, 2230–2237.
- (12) Loehman, R.; Rao, C. R.; Honig, J. M. Crystallography and defect chemistry of solid solutions of vanadium and titanium oxides. *J. Phys. Chem.* **1969**, *73*, 1781–1784.
- (13) Rozier, P.; Ratuszna, A.; Galy, J. Comparative Structural and Electrical Studies of V₂O₃ and V_{2-x}Ni_xO₃ (0<x<0.75) Solid Solution. *Z. Anorg. Allg. Chem.* **2002**, *628*, 1236–1242.
- (14) Ashkenazi, J.; Chuchem, T. Band structure of V₂O₃, and Ti₂O₃. *Phil. Mag.* **1975**, *32*, 763–785.

- (15) Maslen, E. N.; Streltsov, V. A.; Streltsova, N. R.; Ishizawa, N. Synchrotron X-ray study of the electron density in α -Fe₂O₃. *Acta Crystallogr., Sect. B* **1994**, *50*, 435–441.
- (16) Satoshi Sasaki, K. F.; Takeuchi, Y. X-Ray Determination of Electron Density Distributions in Oxides, MgO, MnO, CoO, and NiO, and Atomic Scattering Factors of their Constituent Atoms. *Proc. Jpn. Acad., Ser. B* **1979**, *55*, 43–48.
- (17) Kuriyama, M.; Hosoya, S. X-ray measurement of scattering factors of manganese and oxygen atoms in manganous oxide. *J. Phys. Soc. Jpn.* **1962**, *17*, 1022–1029.
- (18) Jauch, W.; Reehuis, M.; Bleif, H. J.; Kubanek, F.; Pattison, P. Crystallographic symmetry and magnetic structure of CoO. *Phys. Rev. B* **2001**, *64*.
- (19) Shen, Z. X.; List, R. S.; Dessau, D. S.; Wells, B. O.; Jepsen, O.; Arko, A. J.; Barttlet, R.; Shih, C. K.; Parmigiani, F.; Huang, J. C.; Lindberg, P. A. Electronic structure of NiO: Correlation and band effects. *Phys. Rev. B* **1991**, *44*, 3604–3626.
- (20) Fjellvåg, H.; Grønvold, F.; Stølen, S.; Hauback, B. On the crystallographic and magnetic structures of nearly stoichiometric iron monoxide. *J. Solid State Chem.* **1996**, *124*, 52–57.
- (21) Hartmann, H.; Mässung, W. Elektrolyse in Phosphatschmelzen. IV. Über die Elektrolyse von Vanadin-, Niob-und Tantaloxyd in Phosphatschmelzen. *Z. Anorg. Allg. Chem.* **1951**, *266*, 98–104.
- (22) Boucher, B.; Buhl, R.; Perrin, M. Magnetic structure of Mn₃O₄ by neutron diffraction. *J. Appl. Phys.* **1971**, *42*, 1615–1617.
- (23) Gautam, G. S.; Carter, E. A. Evaluating transition metal oxides within DFT-SCAN and SCAN+U frameworks for solar thermochemical applications. *Phys. Rev. Mater.* **2018**, *2*, 095401.
- (24) Abrahams, S.; Calhoun, B. The low-temperature transition in magnetite. *Acta Crystallographica* **1953**, *6*, 105–106.
- (25) Iizumi, M.; Koetzle, T.; Shirane, G.; Chikazumi, S.; Matsui, M.; Todo, S. Structure of magnetite (Fe₃O₄) below the Verwey transition temperature. *Acta Crystallogr., Sect. B* **1982**, *38*, 2121–2133.
- (26) Verwey, E. J. W. Electronic conduction of magnetite (Fe₃O₄) and its transition point at low temperatures. *Nature* **1939**, *144*, 327–328.

- (27) Park, S. K.; Ishikawa, T.; Tokura, Y. Charge-gap formation upon the Verwey transition in Fe_3O_4 . *Phys. Rev. B* **1998**, *58*, 3717.
- (28) Smith, W. L.; Hobson, A. D. The structure of cobalt oxide, Co_3O_4 . *Acta Crystallogr., Sect. B* **1973**, *29*, 362–363.
- (29) Knop, O.; Reid, K.; Sutarno,; Nakagawa, Y. Chalkogenides of the transition elements. VI. X-Ray, neutron, and magnetic investigation of the spinels Co_3O_4 , NiCo_2O_4 , Co_3S_4 , and NiCo_2S_4 . *Can. J. Chem.* **1968**, *46*, 3463.
- (30) Roth, W. L. The magnetic structure of Co_3O_4 . *J. Phys. Chem. Solids* **1964**, *25*, 1–10.
- (31) Cockayne, E.; Levin, I.; Wu, H.; Llobet, A. Magnetic structure of bixbyite $\alpha\text{-Mn}_2\text{O}_3$: A combined DFT+ U and neutron diffraction study. *Phys. Rev. B* **2013**, *87*, 184413.
- (32) Wang, M.; Navrotsky, A. Enthalpy of formation of LiNiO_2 , LiCoO_2 and their solid solution, $\text{LiNi}_{1-x}\text{Co}_x\text{O}_2$. *Solid State Ion.* **2004**, *166*, 167–173.
- (33) Enjalbert, R.; Galy, J. A refinement of the structure of V_2O_5 . *Acta Crystallogr., Sect. C* **1986**, *42*, 1467–1469.
- (34) Ghijsen, J.; Tjeng, L. H.; Elp, J. V.; Eskes, H.; Westerink, J.; Sawatzky, G. A.; Czyzyk, M. T. Electronic structure of Cu_2O and CuO . *Phys. Rev. B* **1988**, *38*, 11322–11330.
- (35) Mittal, R.; Chaplot, S. L.; Mishra, S. K.; Bose, P. P. Inelastic neutron scattering and lattice dynamical calculation of negative thermal expansion compounds Cu_2O and Ag_2O . *Phys. Rev. B* **2007**, *75*, 174303.
- (36) Long, O. Y.; Gautam, G. S.; Carter, E. A. Evaluating optimal U for 3d transition-metal oxides within the SCAN+ U framework. *Phys. Rev. Mater.* **2020**, *4*, 045401.
- (37) Moon, R. M.; Riste, T.; Koehler, W. C.; Abrahams, S. C. Absence of Antiferromagnetism in Ti_2O_3 . *J. Appl. Phys.* **1969**, *40*, 1445–1447.
- (38) Uchida, M.; Fujioka, J.; Onose, Y.; Tokura, Y. Charge Dynamics in Thermally and Doping Induced Insulator-Metal Transitions of $(\text{Ti}_{1-x}\text{V}_x)_2\text{O}_3$. *Phys. Rev. Lett.* **2008**, *101*, 066406.
- (39) Serpone, N. Is the band gap of pristine TiO_2 narrowed by anion-and cation-doping of titanium dioxide in second-generation photocatalysts? *J. Phys. Chem. B* **2006**, *110*, 24287–24293.
- (40) Moon, R. M. Antiferromagnetism in V_2O_3 . *J. Appl. Phys.* **1970**, *41*, 883–883.

- (41) Shin, S.; Tezuka, Y.; Kinoshita, T.; Kakizaki, A.; Ishii, T.; Ueda, Y.; Jang, W.; Takei, H.; Chiba, Y.; Ishigame, M. Observation of local magnetic moments in the Mott transition of V₂O₃ by means of 3s photoemission. *Phys. Rev. B* **1992**, *46*, 9224–9227.
- (42) Shin, S.; Suga, S.; Taniguchi, M.; Fujisawa, M.; Kanzaki, H.; Fujimori, A.; Daimon, H.; Ueda, Y.; Kosuge, K.; Kachi, S. Vacuum-ultraviolet reflectance and photoemission study of the metal-insulator phase transitions in VO₂, V₆O₁₃, and V₂O₃. *Phys. Rev. B* **1990**, *41*, 4993–5009.
- (43) Leroux-Hugon, P.; Paquet, D. Magnetic Susceptibility in the Insulating And Metallic Phases of VO₂: The Contribution of Electron-Electron AND Electron-Lattice Interactions. *J.Phys. Colloq.* **1980**, *41*, C5–67–C5–70.
- (44) Kumar, A.; Singh, P.; Kulkarni, N.; Kaur, D. Structural and optical studies of nanocrystalline V₂O₅ thin films. *Thin Solid Films* **2008**, *516*, 912–918.
- (45) Corliss, L. M.; Hastings, J. M.; Nathans, R.; Shirane, G. Magnetic structure of Cr₂O₃. *J. Appl. Phys.* **1965**, *36*, 1099–1100.
- (46) Abdullah, M. M.; Rajab, F. M.; Al-Abbas, S. M. Structural and optical characterization of Cr₂O₃ nanostructures: Evaluation of its dielectric properties. *AIP Adv.* **2014**, *4*, 027121.
- (47) Misho, R. H.; Murad, W. A.; Fattahallah, G. H. Preparation and optical properties of thin films of CrO₃ and Cr₂O₃ prepared by the method of chemical spray pyrolysis. *Thin Solid Films* **1989**, *169*, 235–239.
- (48) Cheetham, A. K.; Hope, D. A. O. Magnetic ordering and exchange effects in the antiferromagnetic solid solutions Mn_xNi_{1-x}O. *Phys. Rev. B* **1983**, *27*, 6964–6967.
- (49) Messick, L.; Walker, W.; Glosser, R. Direct and temperature-modulated reflectance spectra of MnO, CoO, and NiO. *Phys. Rev. B* **1972**, *6*, 3941–3949.
- (50) Jensen, G. B.; Nielsen, O. The magnetic structure of Mn₃O₄ Hausmannite between 4.7K and Neel point, 41K. *J. Solid State Phys.* **1974**, *7*, 409.
- (51) Xu, H. Y.; Le Xu, S.; Li, X. D.; Wang, H.; Yan, H. Chemical bath deposition of hausmannite Mn₃O₄ thin films. *Appl. Surf. Sci.* **2006**, *252*, 4091–4096.
- (52) Regulski, M.; Przeniosło, R.; Sosnowska, I.; Hohlwein, D.; Schneider, R. Neutron diffraction study of the magnetic structure of α -Mn₂O₃. *J. Alloys Compd.* **2004**, *362*, 236–240.

- (53) Rahaman, H.; Laha, R. M.; Maiti, D. K.; Ghosh, S. K. Fabrication of Mn₂O₃ nanorods: an efficient catalyst for selective transformation of alcohols to aldehydes. *RSC Adv.* **2015**, *5*, 33923–33929.
- (54) Javed, Q.-u.-a.; Feng-Ping, W.; Rafique, M. Y.; Toufiq, A. M.; Iqbal, M. Z. Canted antiferromagnetic and optical properties of nanostructures of Mn₂O₃ prepared by hydrothermal synthesis. *Chin. Phys. B* **2012**, *21*, 117311.
- (55) Druilhe, R.; Suchet, J. P. Electron transport in CrO₂ and Mn_xCr_{1-x}O₂. *Czechoslov. J. Phys. B* **1967**, *17*, 337–346.
- (56) Islam, A. K. M.; Islam, R.; Khan, K. A. Studies on the thermoelectric effect in semiconducting MnO₂ thin films. *J. Mater. Sci.: Mater. Electron.* **2005**, *16*, 203–207.
- (57) Battle, P. D.; Cheetham, A. K. The magnetic structure of non-stoichiometric ferrous oxide. *J. Solid State Phys.* **1979**, *12*, 337.
- (58) Roth, W. L. Magnetic structures of MnO, FeO, CoO, and NiO. *Phys. Rev.* **1958**, *110*, 1333–1341.
- (59) Bowen, H. K.; Adler, D.; Auker, B. H. Electrical and optical properties of FeO. *J. Solid State Chem.* **1975**, *12*, 355–359.
- (60) Wright, J. P.; Attfield, J. P.; Radaelli, P. G. Charge ordered structure of magnetite Fe₃O₄ below the Verwey transition. *Phys. Rev. B* **2002**, *66*, 214422.
- (61) Shull, C. G.; Strauser, W. A.; Wollan, E. O. Neutron diffraction by paramagnetic and antiferromagnetic substances. *Phys. Rev.* **1951**, *83*, 333–345.
- (62) Droubay, T.; Rosso, K. M.; Heald, S. M.; McCREADY, D. E.; Wang, C. M.; Chambers, S. A. Structure, magnetism, and conductivity in epitaxial Ti-doped α -Fe₂O₃ hematite: Experiment and density functional theory calculations. *Phys. Rev. B* **2007**, *75*, 104412.
- (63) Khan, D. C.; Erickson, R. A. Magnetic form factor of Co⁺⁺ ion in cobaltous oxide. *Phys. Rev. B* **1970**, *1*, 2243–2249.
- (64) Van Elp, J.; Wieland, J. L.; Eskes, H.; Kuiper, P.; Sawatzky, G. A.; De Groot, F. M. F.; Turner, T. S. Electronic structure of CoO, Li-doped CoO, and LiCoO₂. *Phys. Rev. B* **1991**, *44*, 6090–6103.
- (65) Wei, P.; Qi, Z. Q. Insulating gap in the transition-metal oxides: A calculation using the local-spin-density approximation with the on-site Coulomb U correlation correction. *Phys. Rev. B* **1994**, *49*, 10864–10868.

- (66) Anisimov, V. I.; Zaanen, J.; Andersen, O. K. Band theory and Mott insulators: Hubbard U instead of Stoner *I. Phys. Rev. B* **1991**, *44*, 943–954.
- (67) Sawatzky, G. A.; Allen, J. W. Magnitude and origin of the band gap in NiO. *Phys. Rev. Lett.* **1984**, *53*, 2339–2342.
- (68) Reynaud, F.; Ghorayeb, A. M.; Ksari, Y.; Menguy, N.; Stepanov, A.; Delmas, C. On the 240 K anomaly in the magnetic properties of LiNiO₂. *Eur. Phys. J. B* **2000**, *14*, 83–90.
- (69) Reynaud, F.; Mertz, D.; Celestini, F.; Debierre, J.-M.; Ghorayeb, A. M.; Simon, P.; Stepanov, A.; Voiron, J.; Delmas, C. Orbital frustration at the origin of the magnetic behavior in LiNiO₂. *Phys. Rev. Lett.* **2001**, *86*, 3638–3641.
- (70) Zhang, L.; McMillon, L.; McNatt, J. Gas-dependent bandgap and electrical conductivity of Cu₂O thin films. *Sol. Energy Mater. Sol. Cells* **2013**, *108*, 230–234.
- (71) Yang, B. X.; Tranquada, J. M.; Shirane, G. Neutron scattering studies of the magnetic structure of cupric oxide. *Phys. Rev. B* **1988**, *38*, 174–178.