

Machine learning guided exploration of amorphous electrodes and electrolytes

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Acknowledgments



Group picture in Jan 2025



Param
Siddhi
(CDAC)



SERC
(IISc)

Pritam
(Poster)

Why amorphous materials?



Compositional and structural flexibility¹

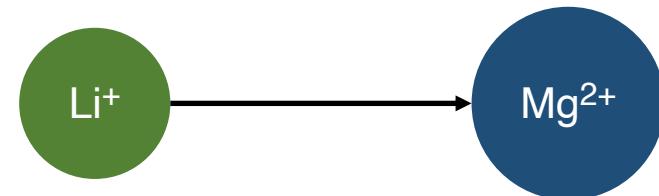
- Electrolytes: significant changes in (local) ionic content without phase transformations
- Electrodes: significant capacities without phase transformations

Electrolytes: can hinder nucleation of dendrites²

- Lack of structural 'inhomogeneities' that cause nucleation
- No grain boundaries

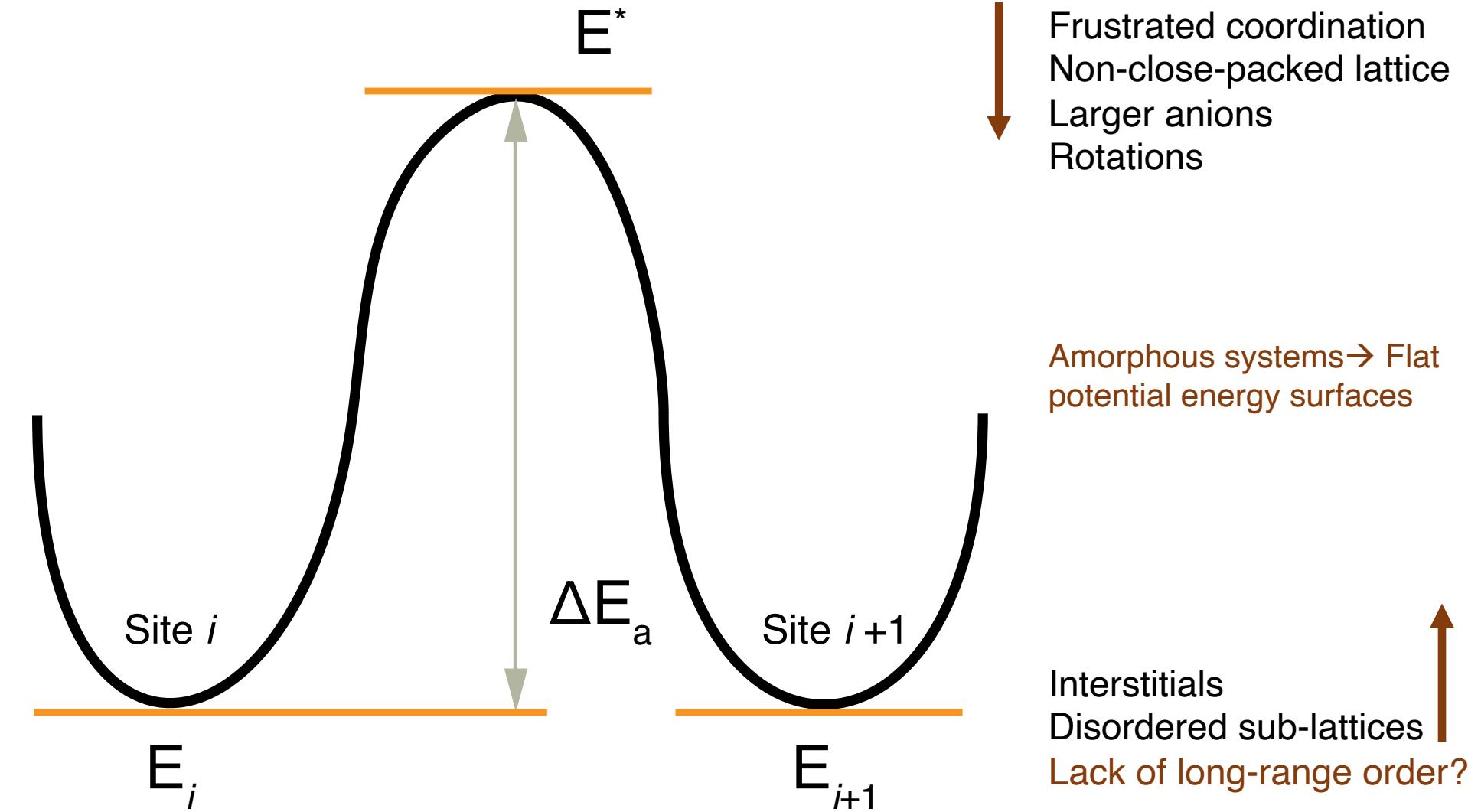
Electrodes: can improve ionic mobility in multivalent systems

- Such as Mg
- 'Flatter' potential energy landscapes



How can amorphous materials improve ionic mobility?*

*Terms and conditions apply



Objectives

Can we computationally model amorphous systems?

- Quantify ionic transport
- Other metrics?

How can we model?

- Use machine learned interatomic potentials (MLIPs)
- Train on density functional theory (DFT) generated data

What systems?

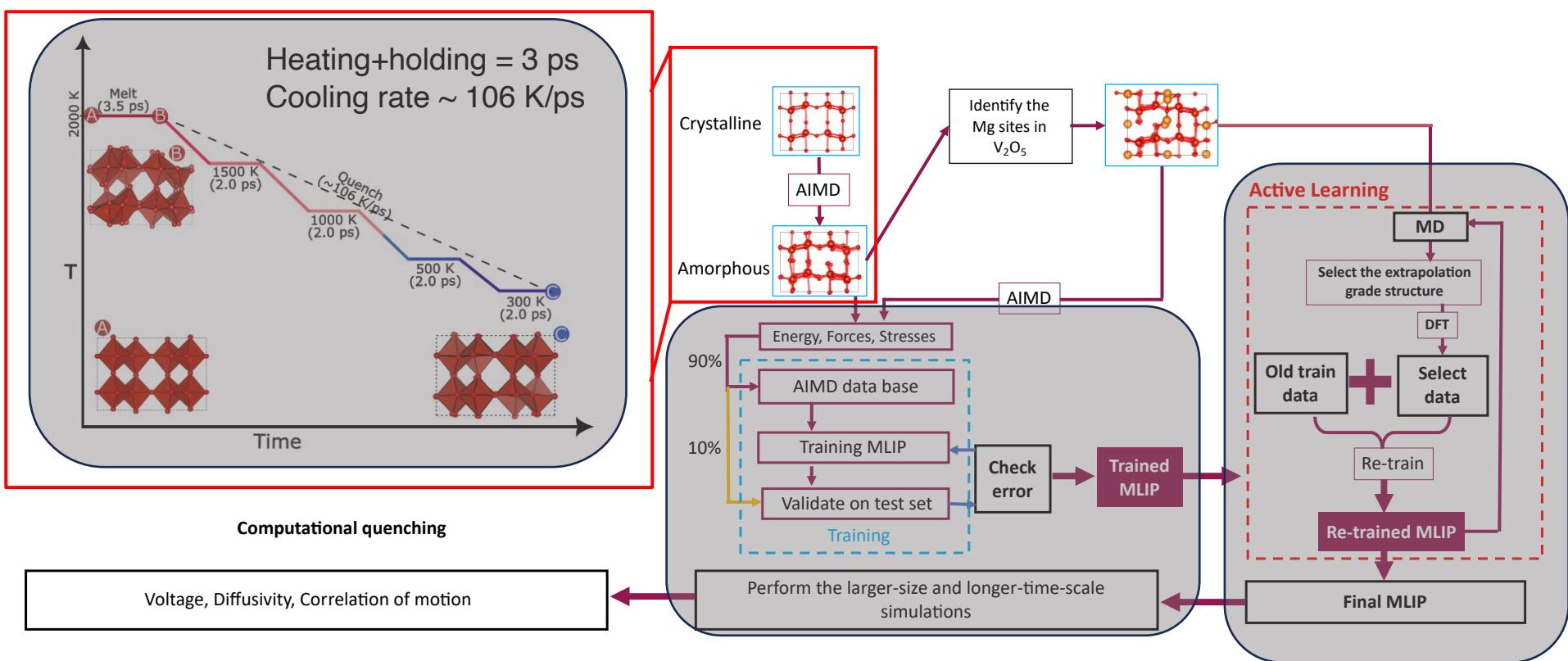
- Lithium phosphorous oxynitride (LiPON) as electrolyte in Li-ion
- V_2O_5 as positive electrode (cathode) in Mg

Quantities/metrics

- LiPON: structure, transport (bulk and interface)
- V_2O_5 : structure, voltages, and transport (bulk)

Workflow

MD: Molecular dynamics
 AIMD: Ab initio MD
 MLIP: Machine learned interatomic potential



LiPON dataset: 13,454 configurations in computed dataset

- Li, Li_3P , Li_3N , Li_3PO_4 , LiPN_2 , Li_2O_2
 - Bulk, strained, melt+quenched, defective, slabs

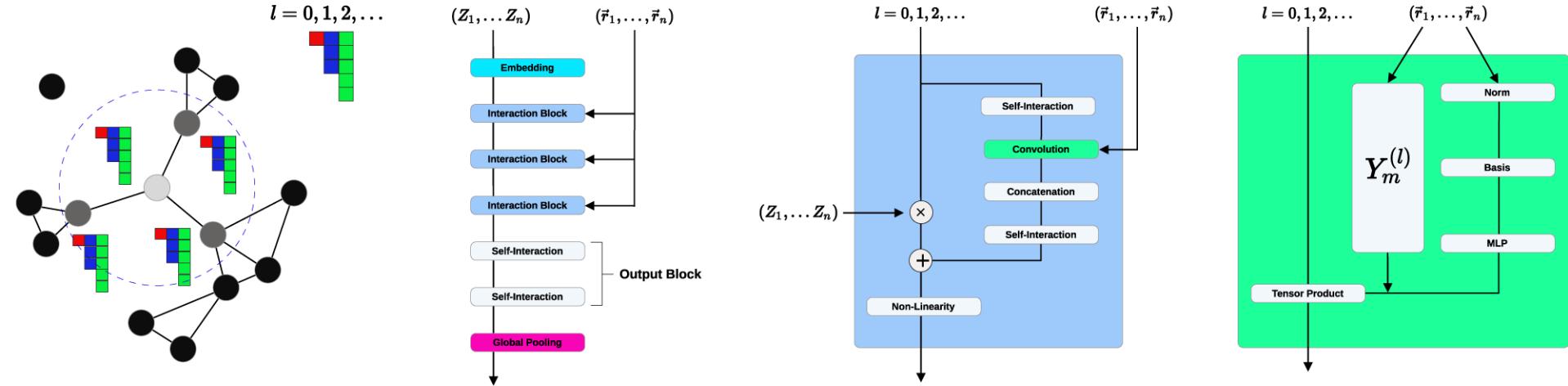
Neural equivariant
 interatomic potential
 (NequiP)

V_2O_5 dataset: 3,725 computed configurations

- Melt+quench (+active learning)
- $\text{V}_2\text{O}_5 + \text{MgV}_2\text{O}_5$

Moment tensor potential
 (MTP)

NequIP for LiPON



Based on deep, graph neural networks

- Nodes (atoms) and edges (bonds)
- Message passing
- Equivariance

Every atom has a feature vector of different orders

- Scalars, vectors, and tensors

Radial and angular basis functions

$$E_{pot} = \sum_{i \in N_{atoms}} E_{i,atomic}$$

$$\vec{F}_i = -\nabla_i E_{pot}$$

$$B(r_{ij}) = \frac{2}{r_c} \frac{\sin(\frac{b\pi}{r_c} r_{ij})}{r_{ij}} f_{env}(r_{ij}, r_c)$$

MTP for V₂O₅

$$E^{\text{mtp}}(\text{cfg}) = \sum_{i=1}^n V(\mathbf{n}_i)$$

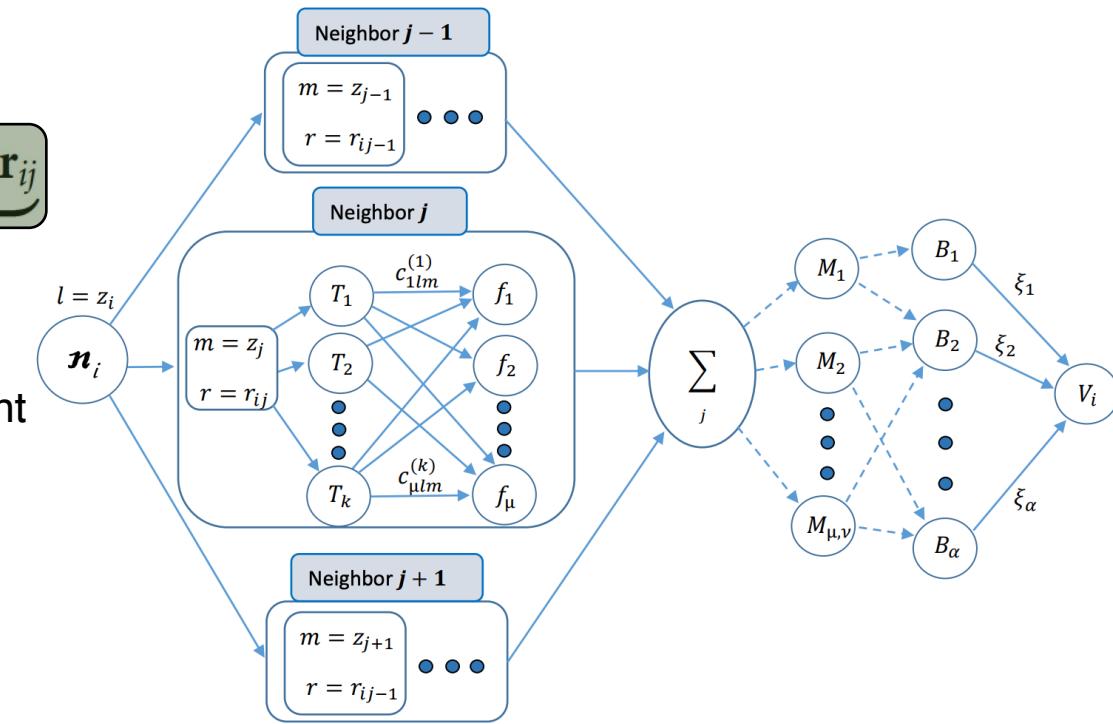
Total energy as a sum of atomic energies

$$V(\mathbf{n}_i) = \sum_{\alpha} \xi_{\alpha} B_{\alpha}(\mathbf{n}_i)$$

Atomic energies – sum of contracted moment tensors capturing many-body interactions

$$M_{\mu,\nu}(\mathbf{n}_i) = \sum_j f_{\mu}(|r_{ij}|, z_i, z_j) \underbrace{\mathbf{r}_{ij} \otimes \dots \otimes \mathbf{r}_{ij}}_{\nu \text{ times}}$$

Radial component
Angular component



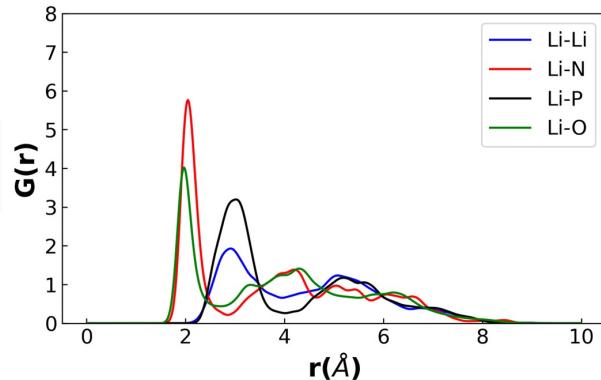
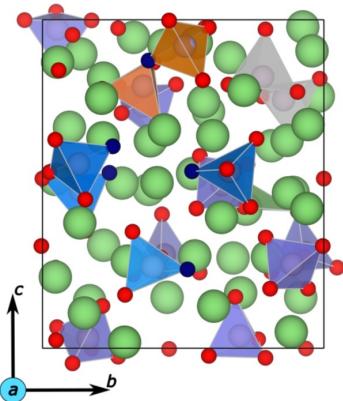
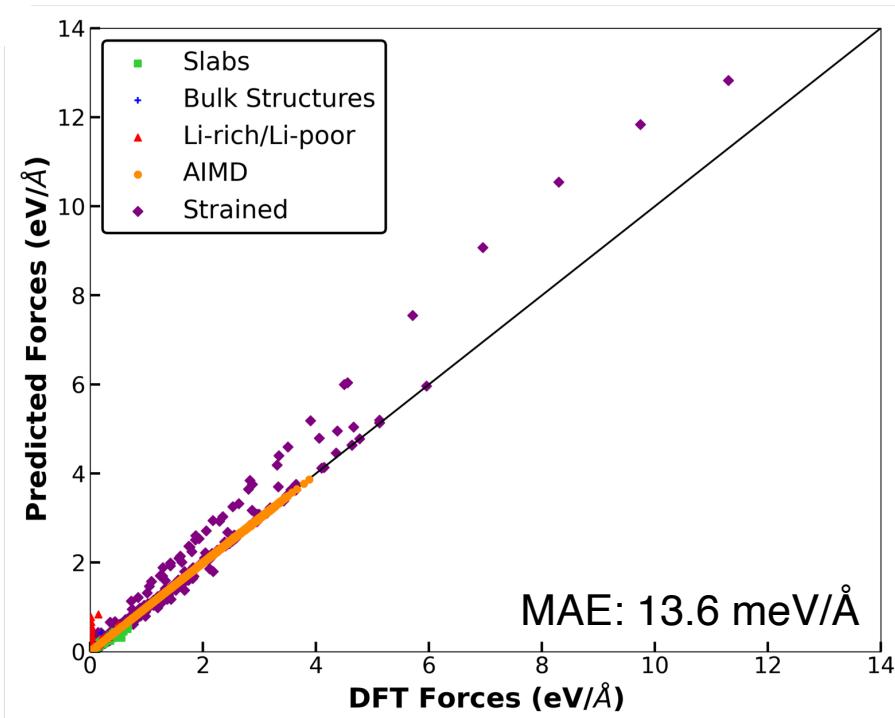
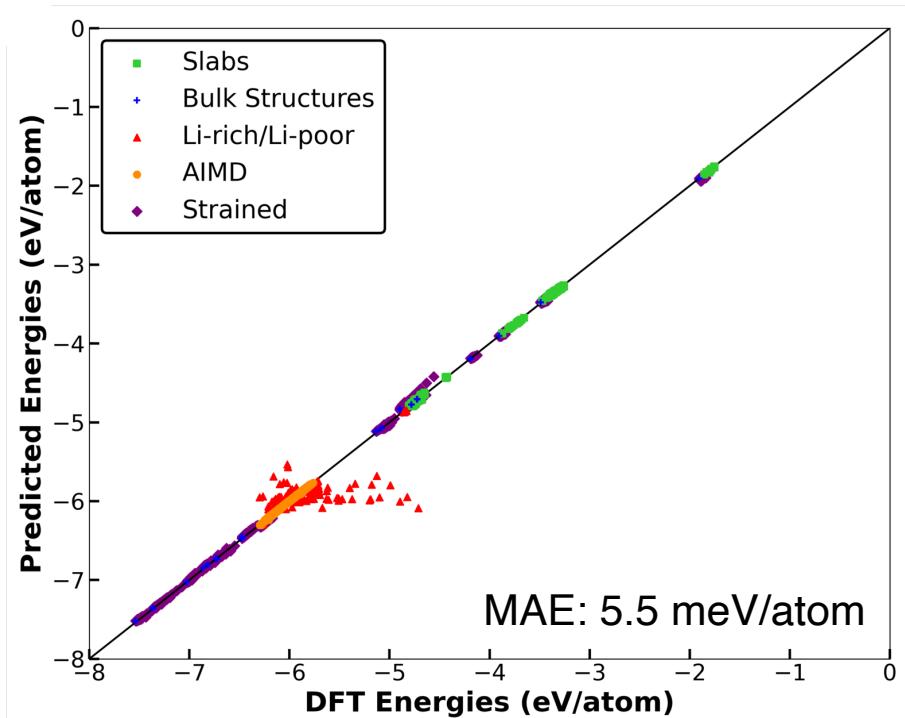
V : function invariant to permutations, rotations, and reflections

- Smooth with respect to exchange of atoms from neighborhood

Results

- Are MLIPs accurate?
- Structures are amorphous?
- LiPON: Bulk and interfacial transport?
- V_2O_5 : What is the loss on voltages?
- V_2O_5 : Amorphous \rightarrow better Mg diffusivity?

LiPON: Accuracy and amorphous structure



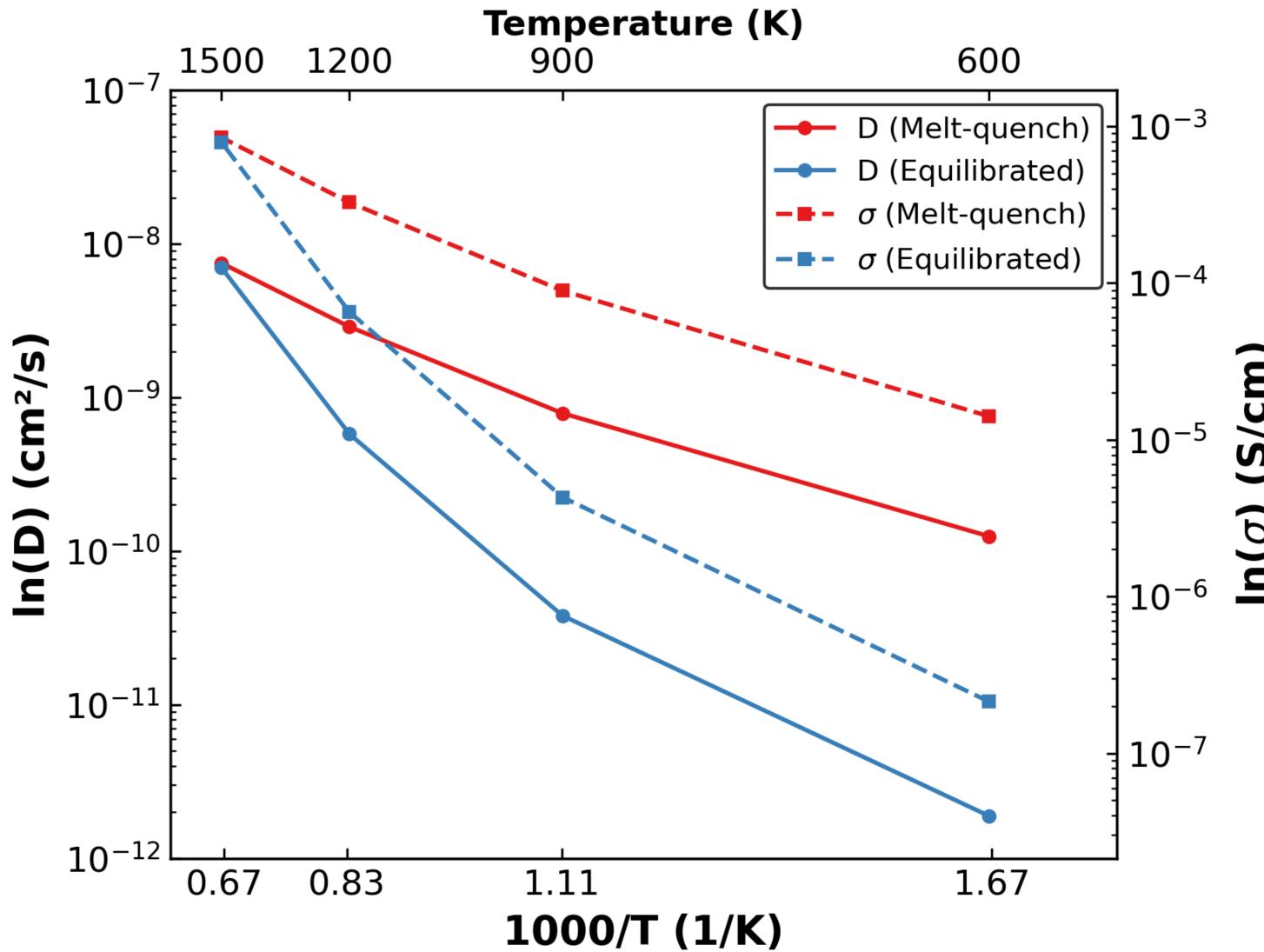
$\text{Li}_{2.94}\text{PO}_{3.5}\text{N}_{0.31}$

- Purple, grey (without N)
- Blue, orange (with N)

Quenched to 250 K

- Broad peaks beyond 4 Å

LiPON: bulk diffusivities higher in structures with higher disorder

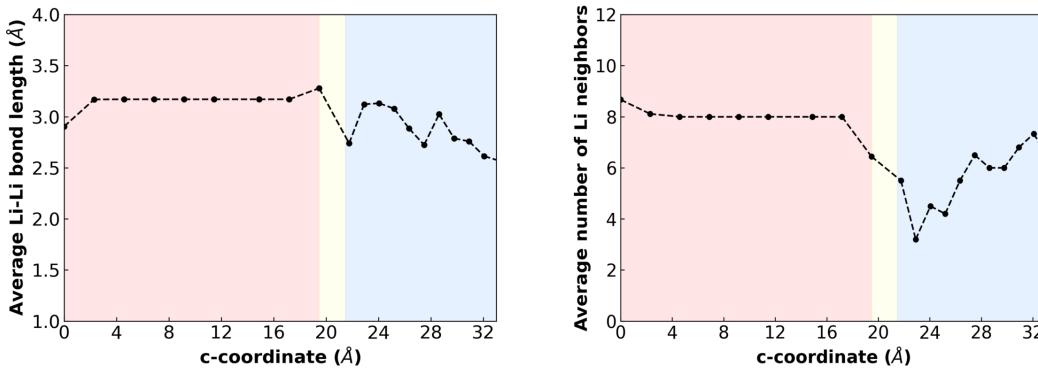
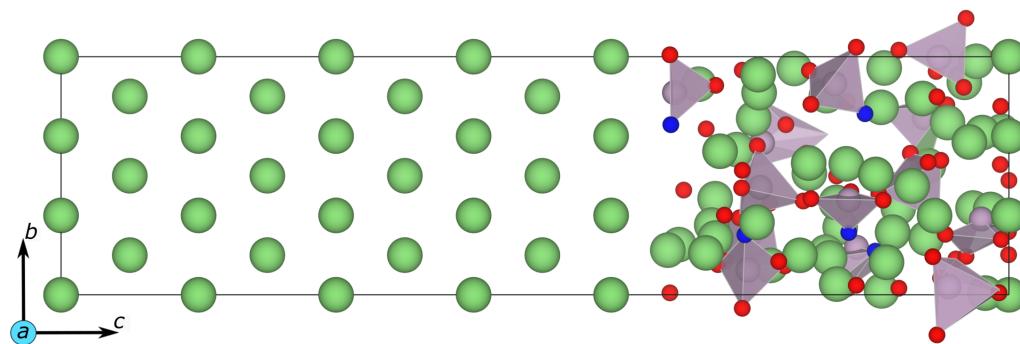


Statistics over a minimum of 100 ps,
124 atoms
• $\text{Li}_{2.94}\text{PO}_{3.5}\text{N}_{0.31}$

Higher orientational disorder \rightarrow flatter landscape

Calculated σ ($\sim 1 \times 10^{-6}$ S/cm) at 300 K matches experiment ($\sim 3 \times 10^{-6}$ S/cm)

Li(110)ILiPON interface

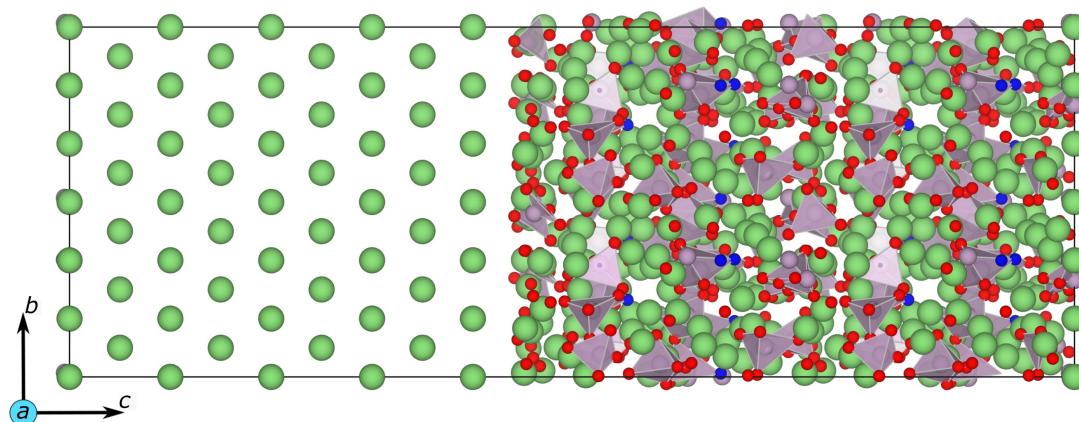


'Smaller' model: ~200 atoms

- 75ps of sampling

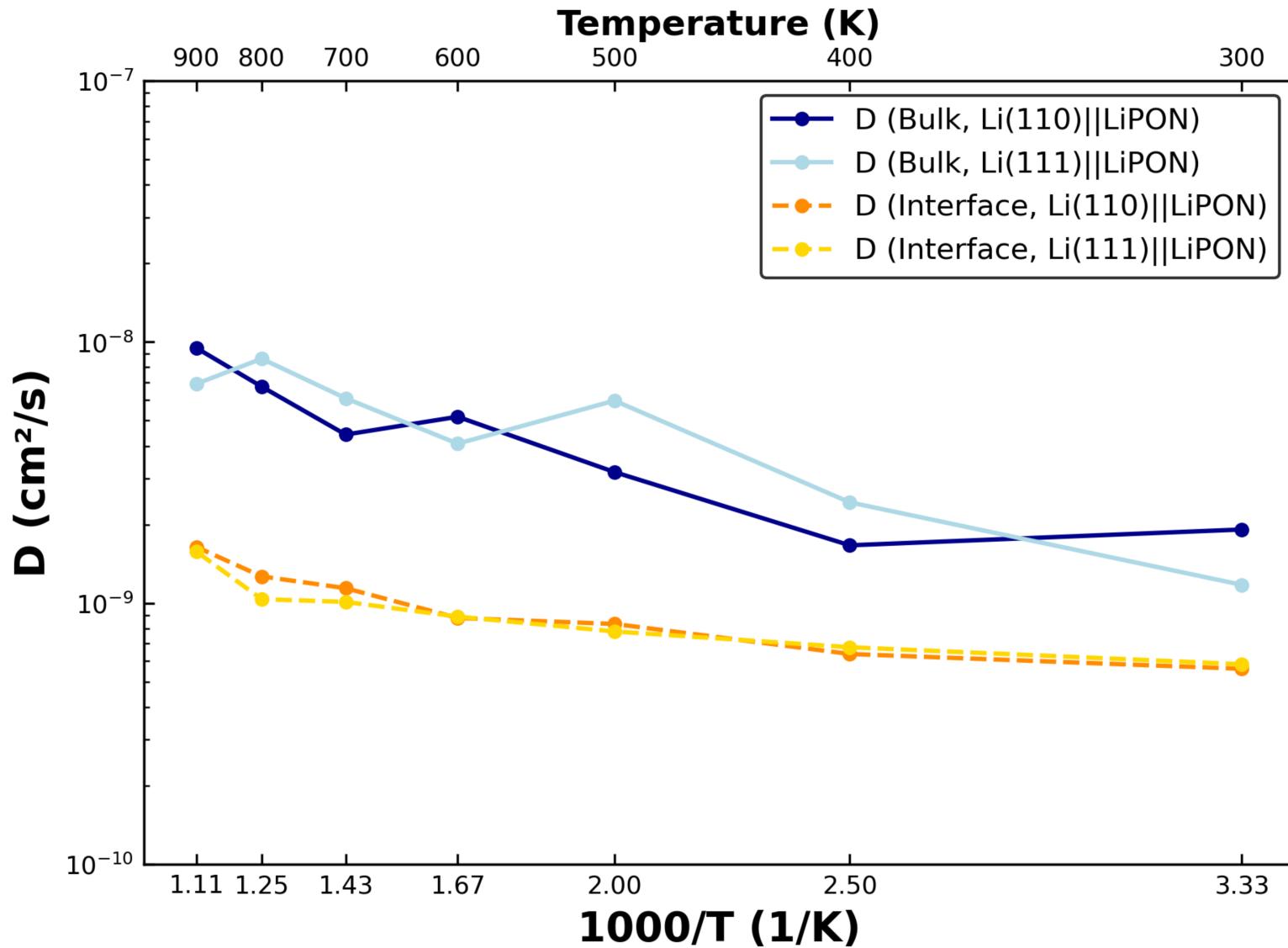
Also Li(111)ILiPON interface

Distribution of Li-neighborhood in the interface model

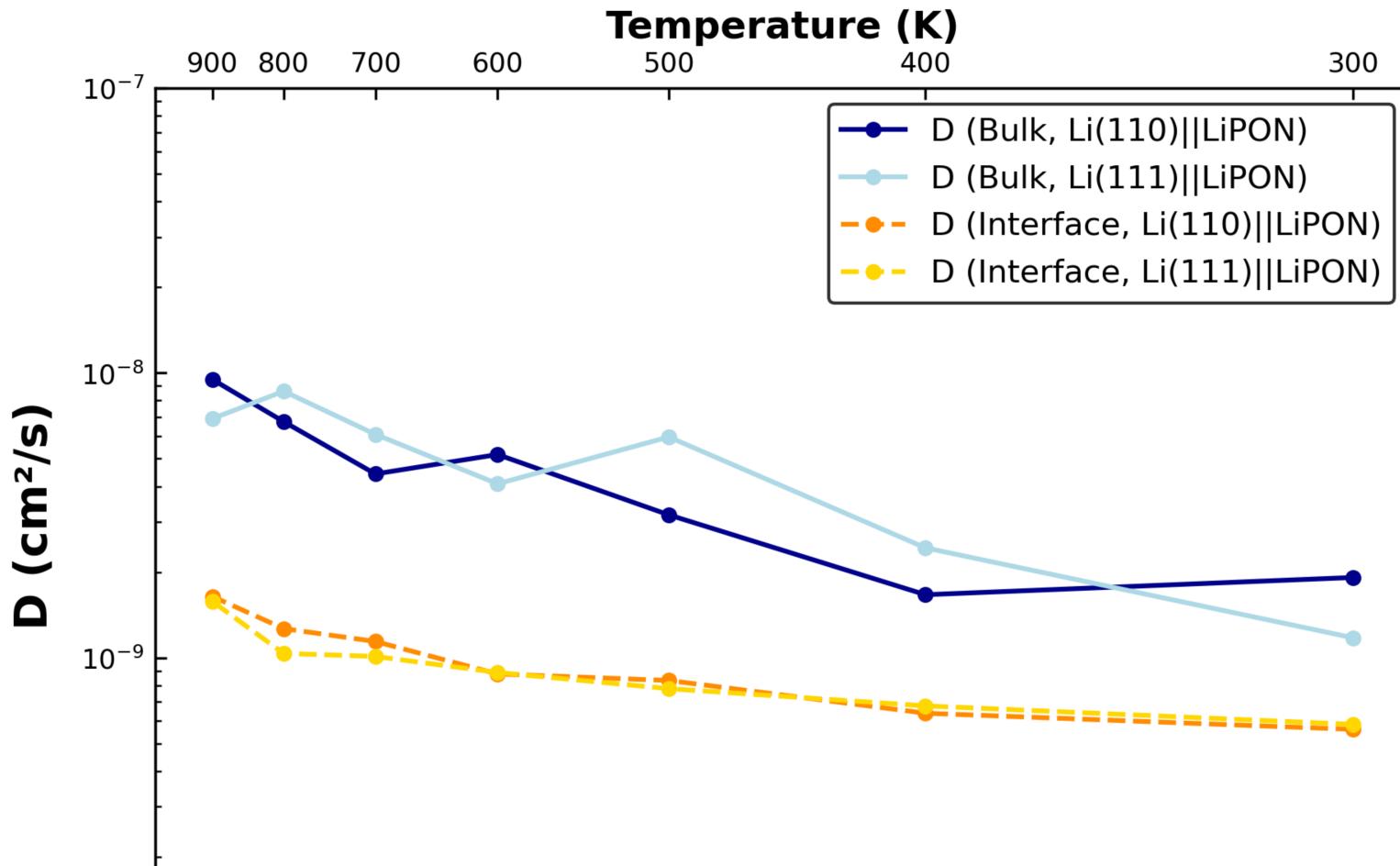


'Larger' model: ~1300 atoms

LiPON: Diffusivity across the interface (marginally) slower than bulk



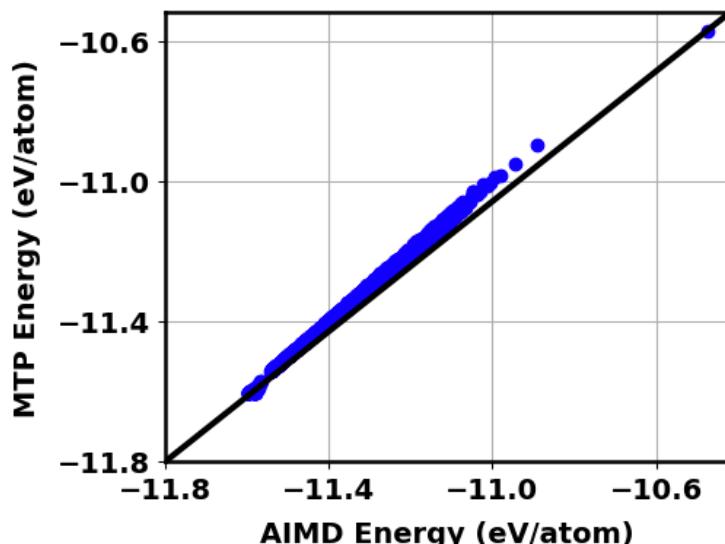
LiPON: Diffusivity across the interface (marginally) slower than bulk



Similarity in Li transport between bulk and interface → mitigate Li-dendrite propagation

V_2O_5 : MTPs are accurate

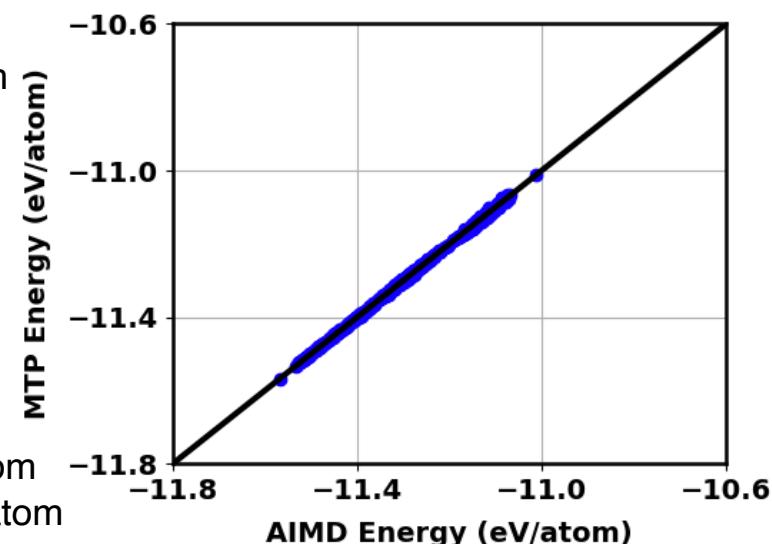
Train



Errors

MAE: 2.97 meV/atom
RMSE: 4.68 meV/atom

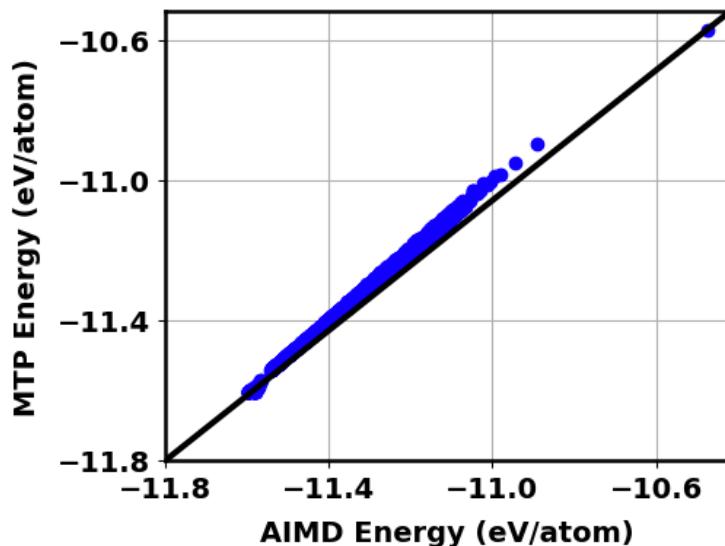
Test



MAE: 2.33 meV/atom
RMSE: 3.16 meV/atom

V_2O_5 : MTPs are accurate

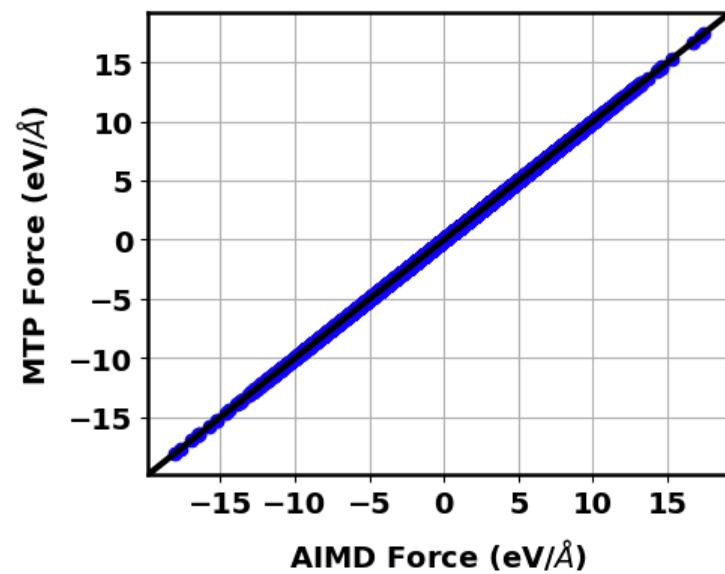
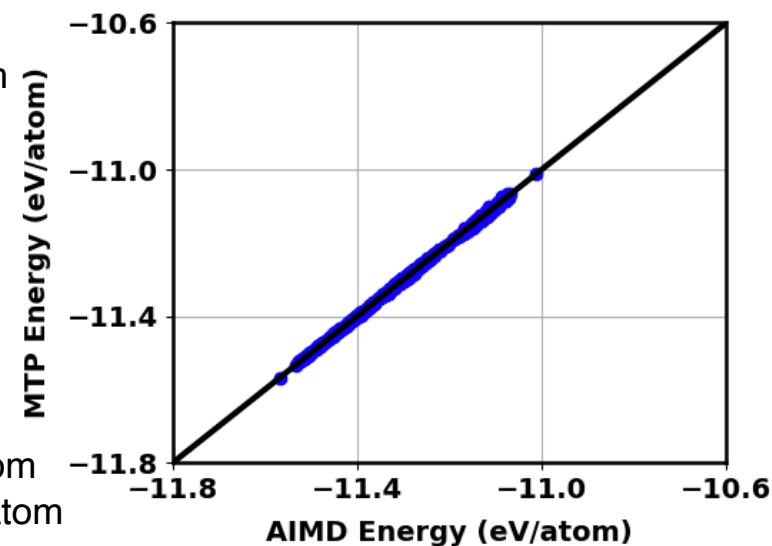
Train



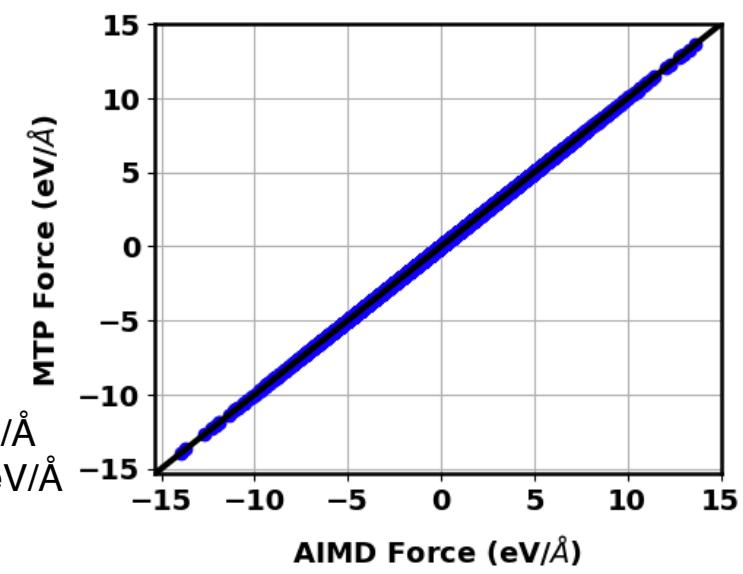
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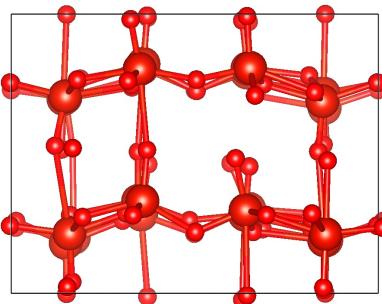
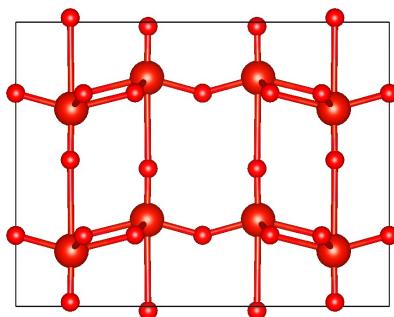
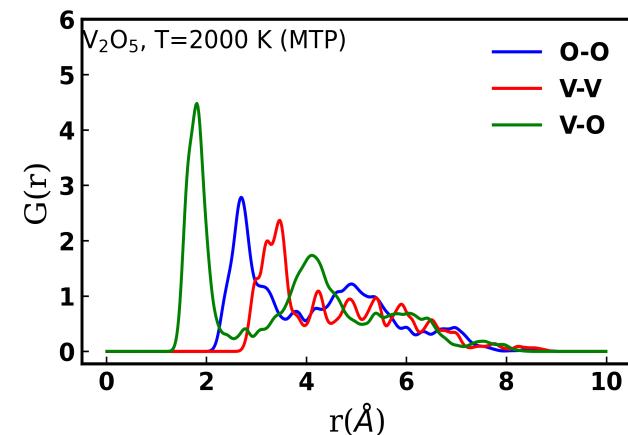
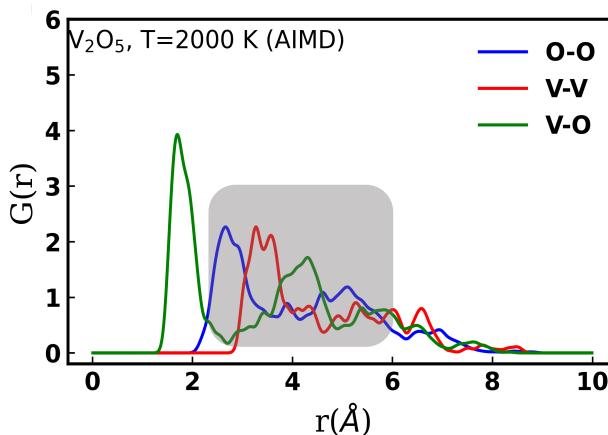
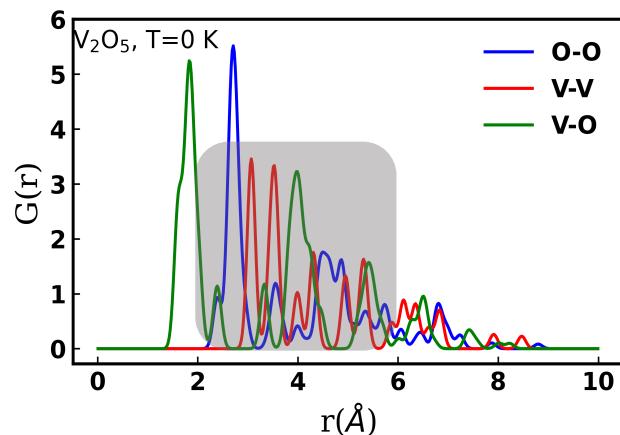


MAE: 0.11 eV/Å
RMSE: 0.24 eV/Å

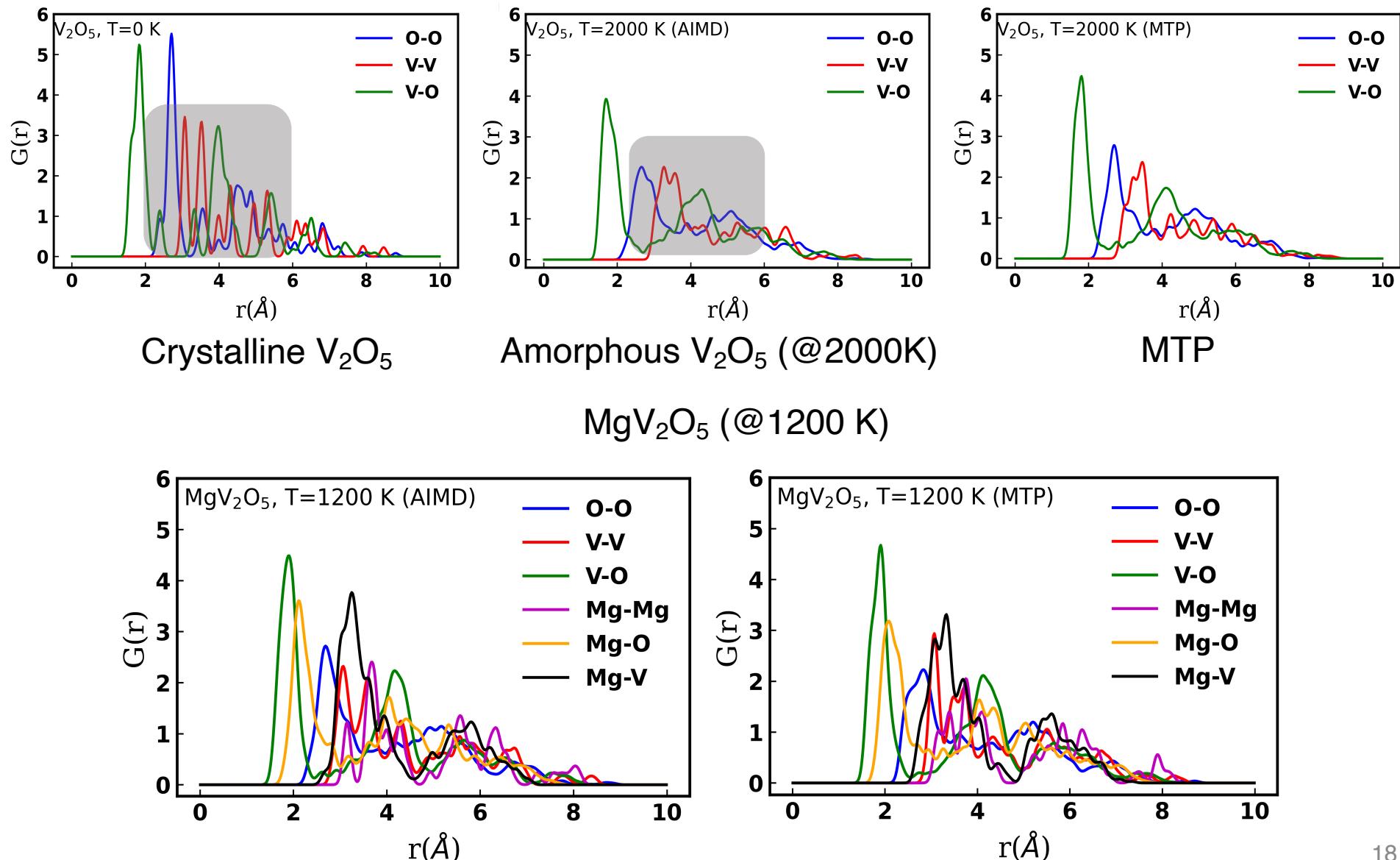


MAE: 0.12 eV/Å
RMSE: 0.24 eV/Å

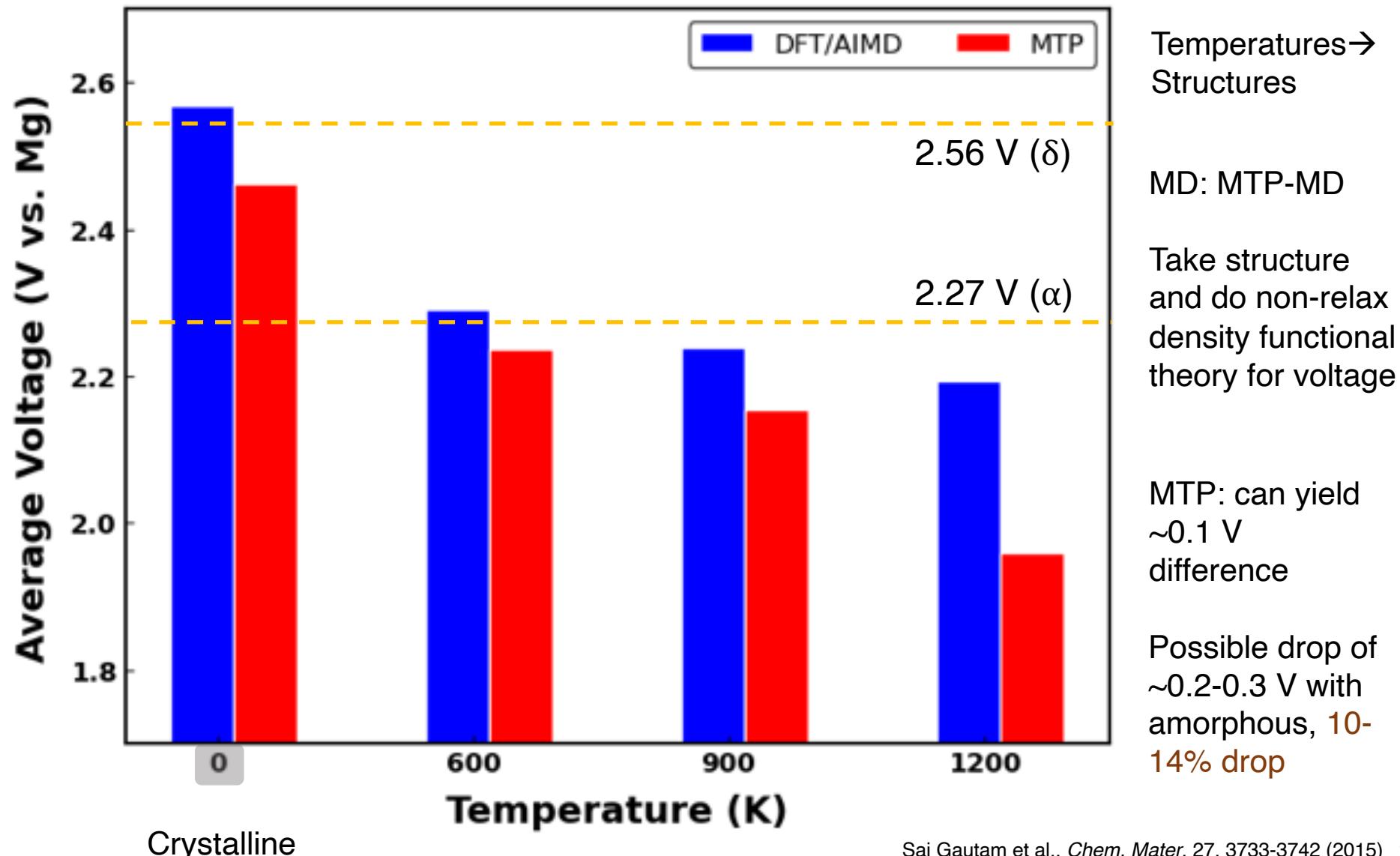
Melt+quench generates (reasonable) amorphous structures



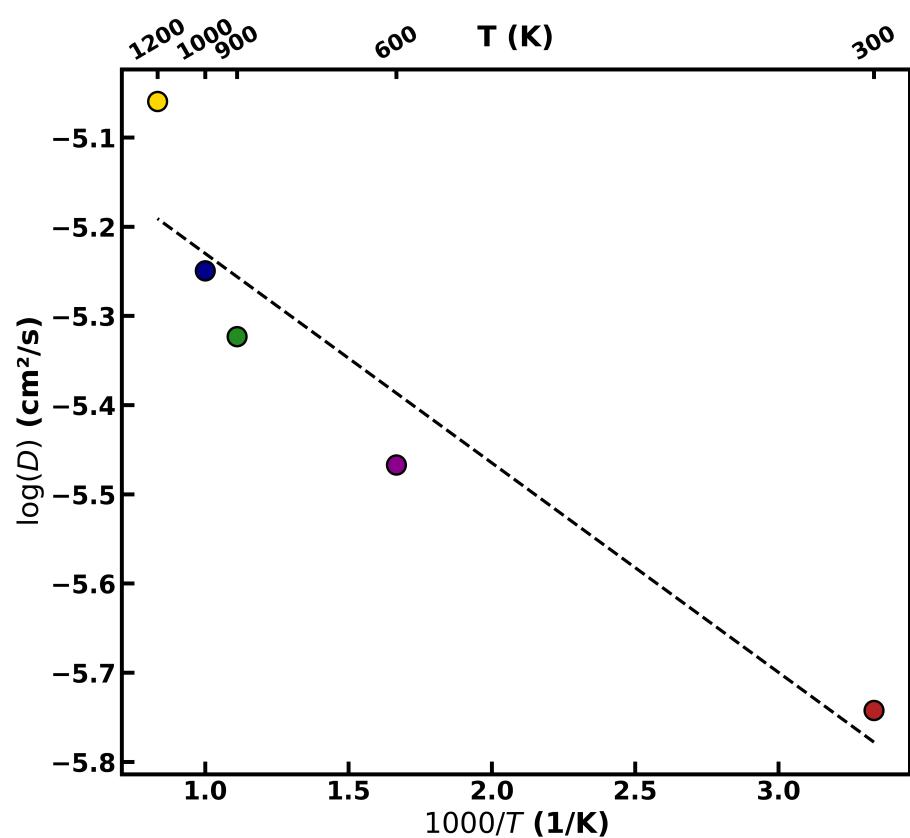
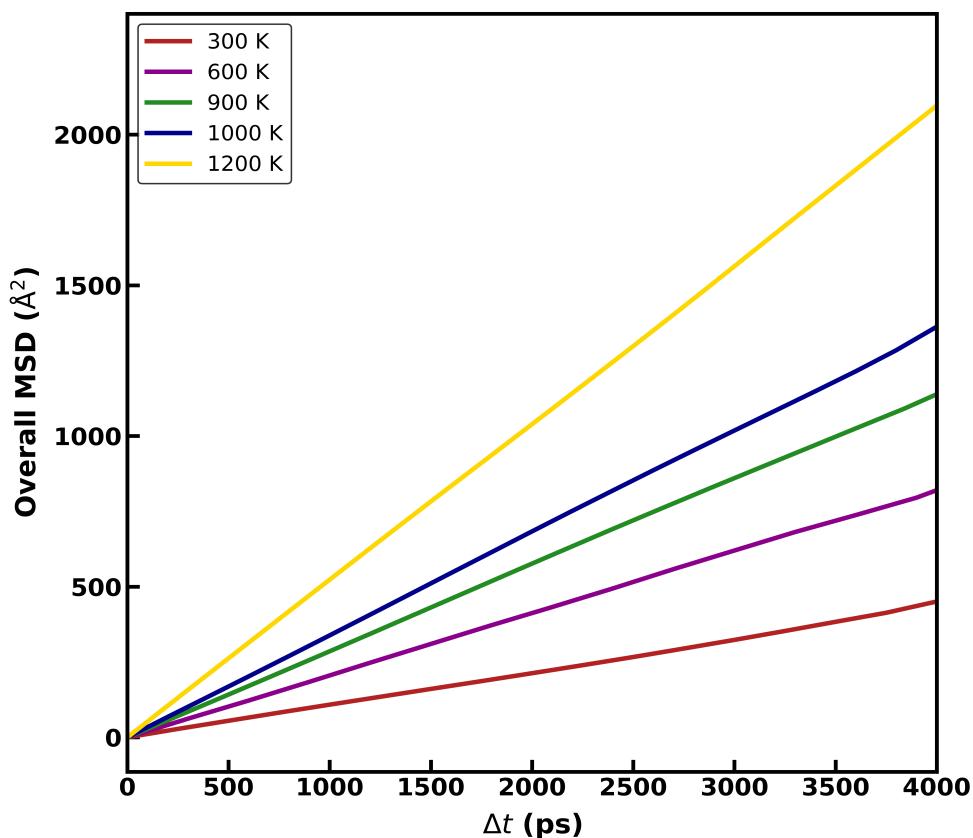
Melt+quench generates (reasonable) amorphous structures



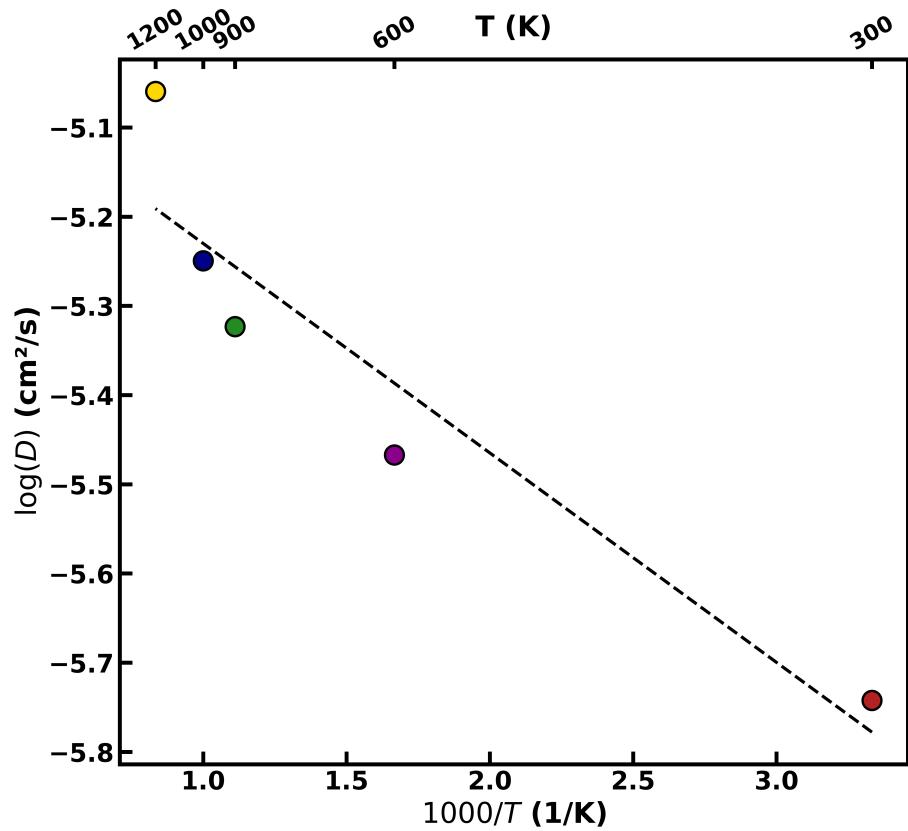
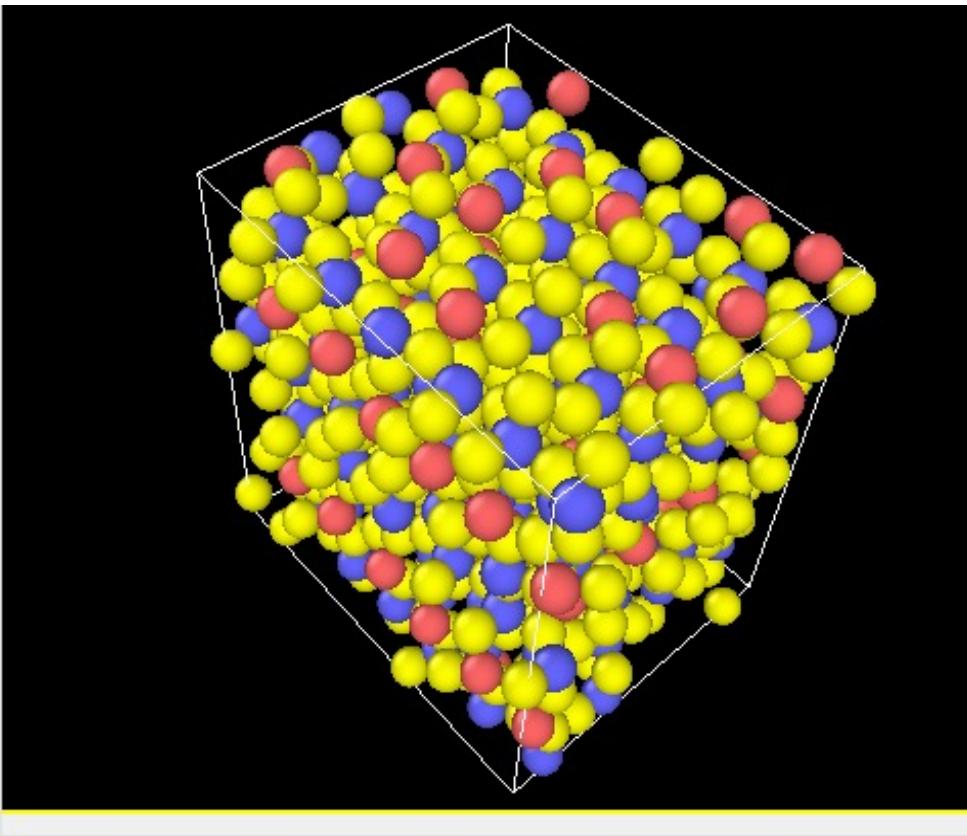
V_2O_5 : Going amorphous doesn't affect voltages (much)



Amorphous V₂O₅ exhibits higher Mg mobility



Amorphous V₂O₅ exhibits higher Mg mobility



MTP; 1ns; 1000 K

Amorphous Diffusivity: $\sim 10^{-6} \text{ cm}^2/\text{s}$; Barrier*: $\sim 46 \text{ meV}$

Crystalline [Chem. Mater. 27, 3733-3742 (2015)]

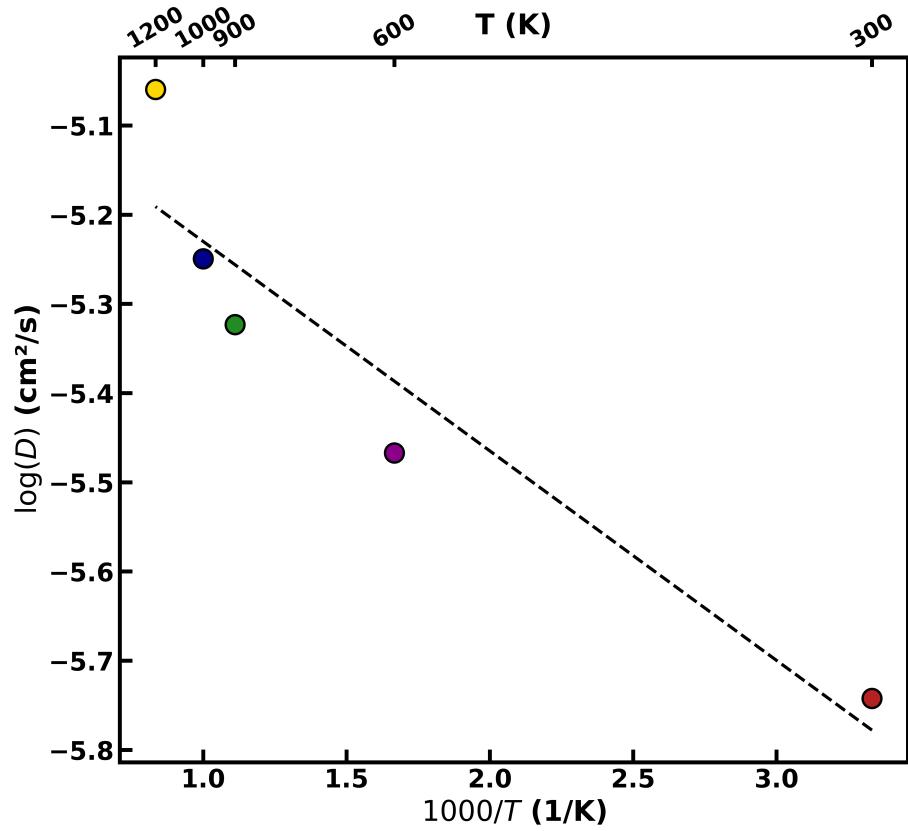
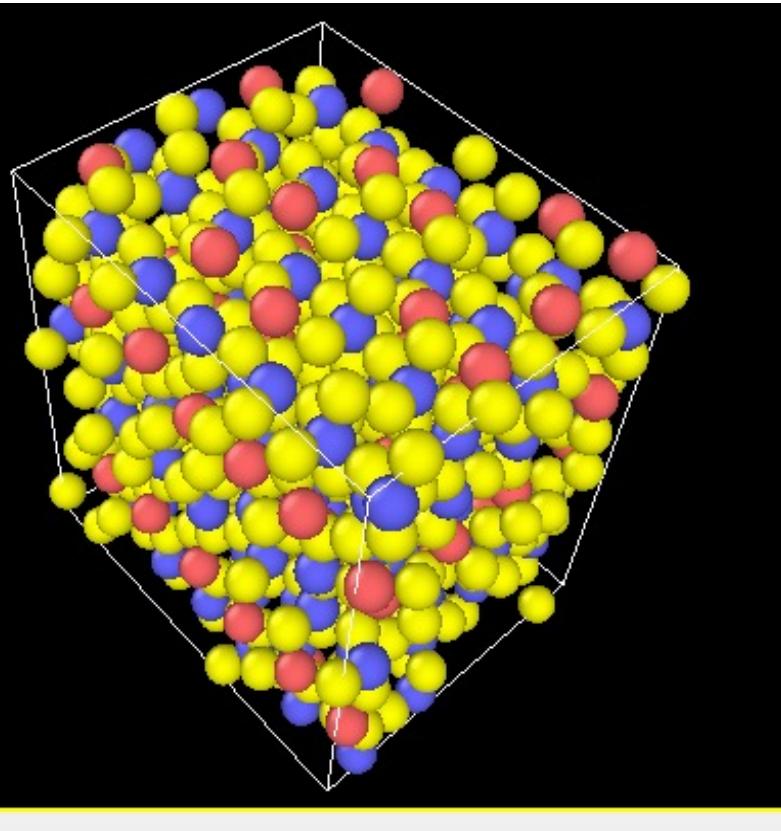
Diffusivity (δ): $\sim 10^{-13}\text{-}10^{-16} \text{ cm}^2/\text{s}$; Barrier: 600-750 meV

Diffusivity (α): $\sim 10^{-20}\text{-}10^{-22} \text{ cm}^2/\text{s}$; Barrier: 975-1120 meV

7 orders of magnitude boost due to amorphization

5 orders of magnitude higher than $\text{Mg}_x\text{Ti}_2\text{S}_4$

Amorphous V₂O₅ exhibits higher Mg mobility



MTP; 1ns; 1000 K

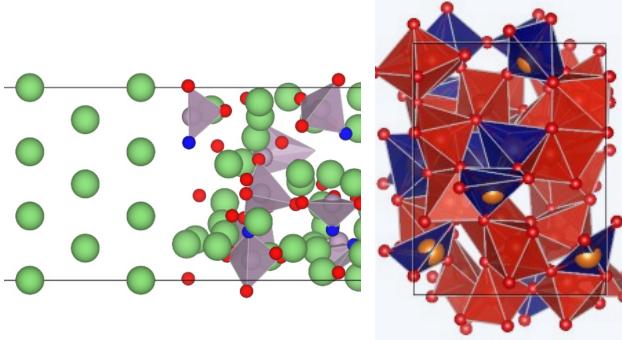
Amorphous Diffusivity: $\sim 10^{-6} \text{ cm}^2/\text{s}$; Barrier*: $\sim 46 \text{ meV}$

7 orders of magnitude boost due to

Can obtain significant diffusivity boost in amorphous (Mg) electrodes with low voltage loss

Conclusions

- Amorphous materials can have use cases in batteries
 - As dendrite-suppressing solid electrolytes in Li
 - As cathodes in Mg
- Use NequIP to model Li||LiPON interface
 - Low diffusivity difference between bulk and interface
- Amorphous V_2O_5 seems reasonable
 - 7 orders-of-magnitude boost in diffusivity



“Investigating Ionic Diffusivity in Amorphous LiPON using Machine-Learned Interatomic Potentials”, A. Seth, R. P. Kulkarni, G. Sai Gautam, *ACS Mater. Au*, 5, 458-468 (2025; ‘Rising Stars 2024’ collection).

“Machine learning guided exploration of amorphous V_2O_5 as cathode for Mg-batteries ”, V. Choyal, D. Dey, and G. Sai Gautam, *arXiv*, 2505.10967 (2025).