

Role of exchange-correlation functionals in migration barrier predictions

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Acknowledgments



Reshma



+CaRe group



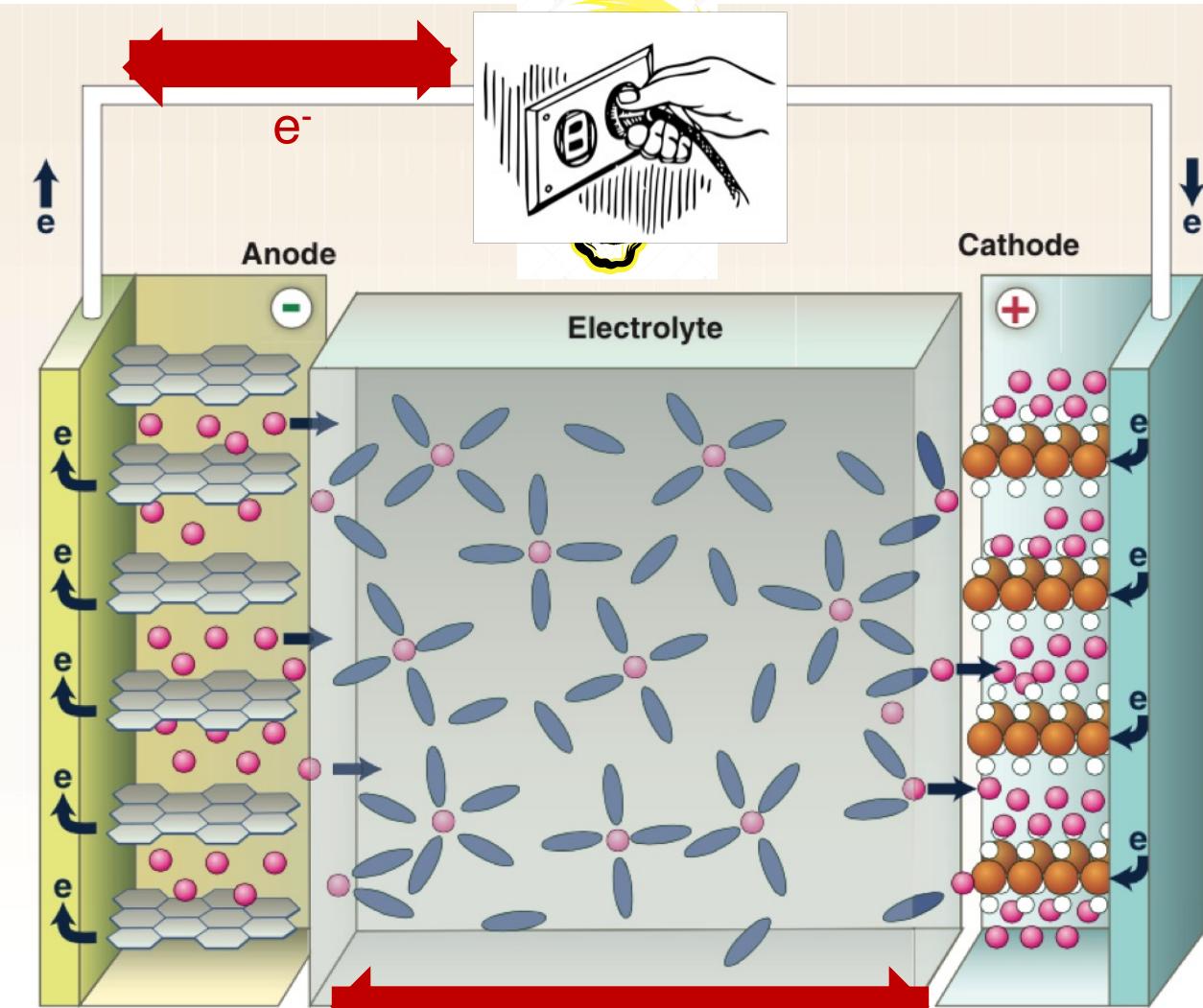
NSCC (Singapore)



SERC (IISc)



How does the modern Li-ion battery work?



Voltage (V): Potential to do work

Capacity (mAh): Amount of charge stored

Voltage*capacity: Energy stored

Rate (C): How fast can a battery be charged and discharged?

- Crucial for power performance

All performance metrics of a battery system are **material dependent**: anode, cathode, and electrolyte

Cu
current
collector

Graphene
structure

Li^+

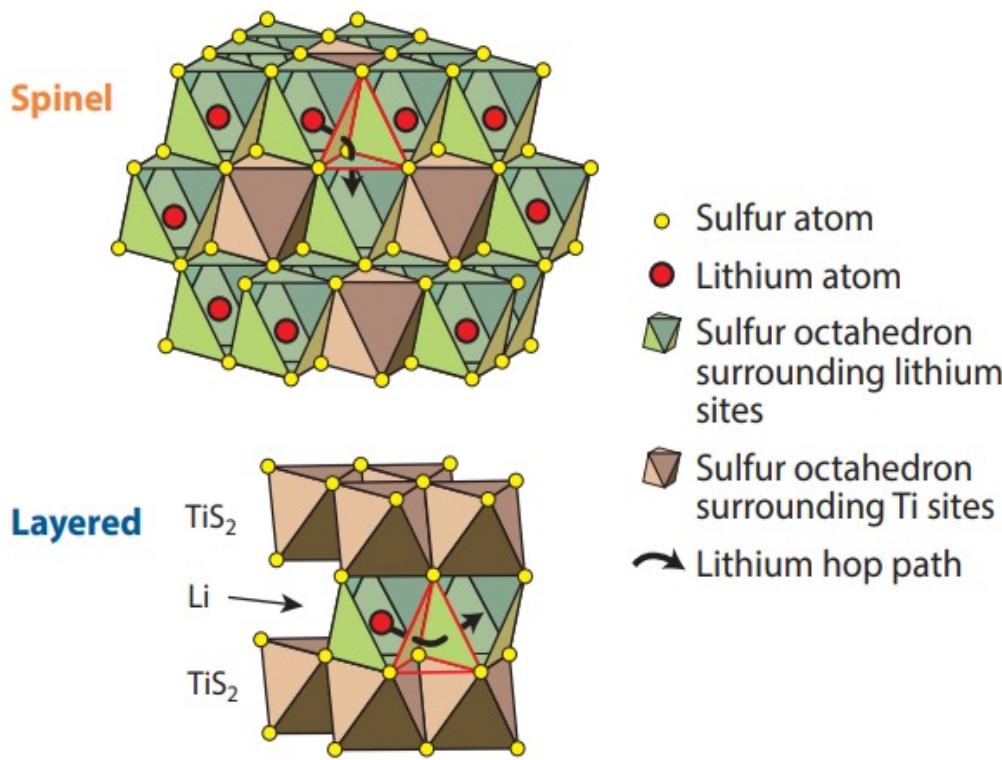
Solvent
molecule

$LiMO_2$ layer
structure

Al
current
collector

Image adopted from B. Dunn et al., Science 2011

Migration barriers govern rate performance in batteries



Intercalation electrodes/solid electrolytes:
ionic diffusivity (D) within the bulk a major factor in rate performance

$$D = D_o \exp\left(-\frac{E_m}{k_B T}\right)$$

D_o : Diffusivity pre-factor (carrier concentration, correlations, etc.)

k_B : Boltzmann constant

T : Temperature

E_m : Migration barrier

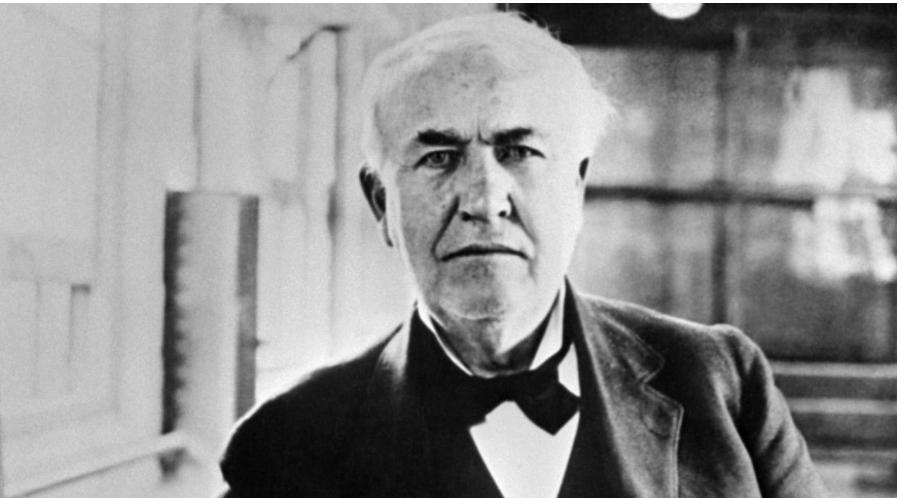
Migration barrier: dominant factor determining diffusivity

Experimental measurements of E_m : variable-temperature impedance spectroscopy, variable temperature nuclear magnetic resonance, etc.

Computational predictions of E_m : ab initio molecular dynamics, nudged elastic band (NEB)

How accurate are computational predictions?

Methods: Edison vs. Iron Man



Trial and error of candidates in a lab

Simulate and identify candidates
(on a transparent touch screen preferably)



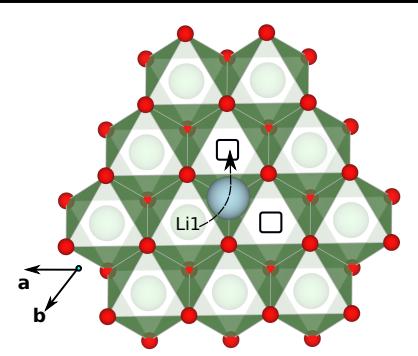
Density functional theory (DFT): [Approximately] predict material properties

- Thermodynamic, kinetic, and electronic properties

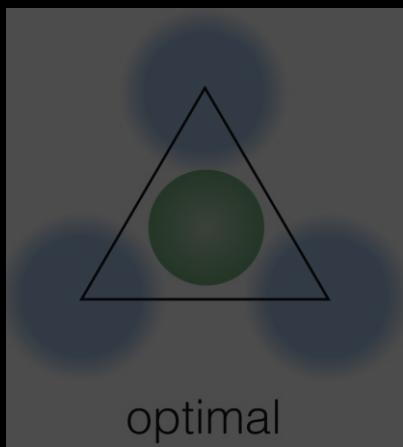
Machine learning: learn from predictions to make better predictions



Objectives

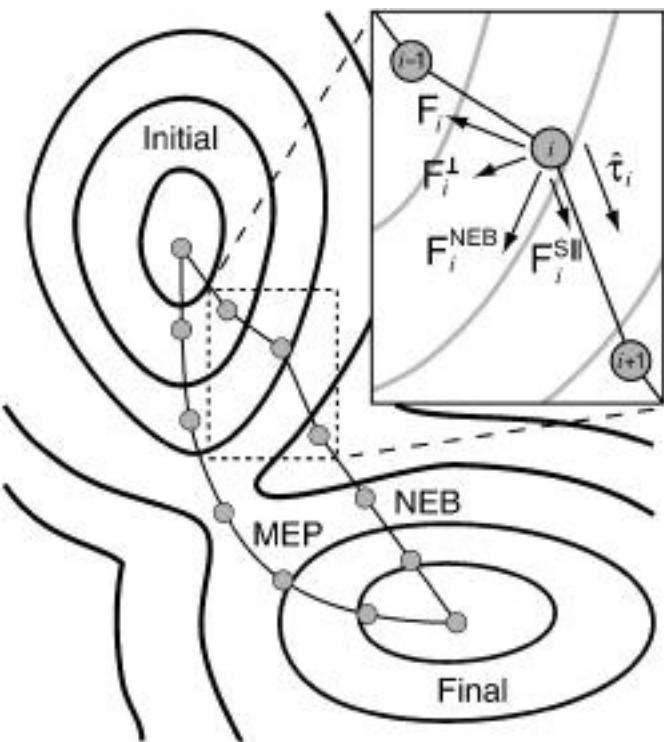


Assessing accuracy of computational predictions of ionic mobility



Examples of cathode discovery in Ca-batteries, including ionic mobility as a criterion

NEB and computational handles for ionic mobility



NEB: can estimate E_m for an ionic hop from one "stable" site to another

Saddle-point finder by optimizing forces "perpendicular" to the potential energy surface

NEB often used with DFT: approximations within DFT affect NEB estimates

Uniform background charge (NE or ne)
[solid electrolytes]

Climbing image (CI)
approximation
[solid electrolytes]

Exchange-correlation (XC) choice

Generalized gradient approximation (GGA)
• Perdew-Burke-Ernzerhof²

Strongly constrained and appropriately normed (SCAN)²

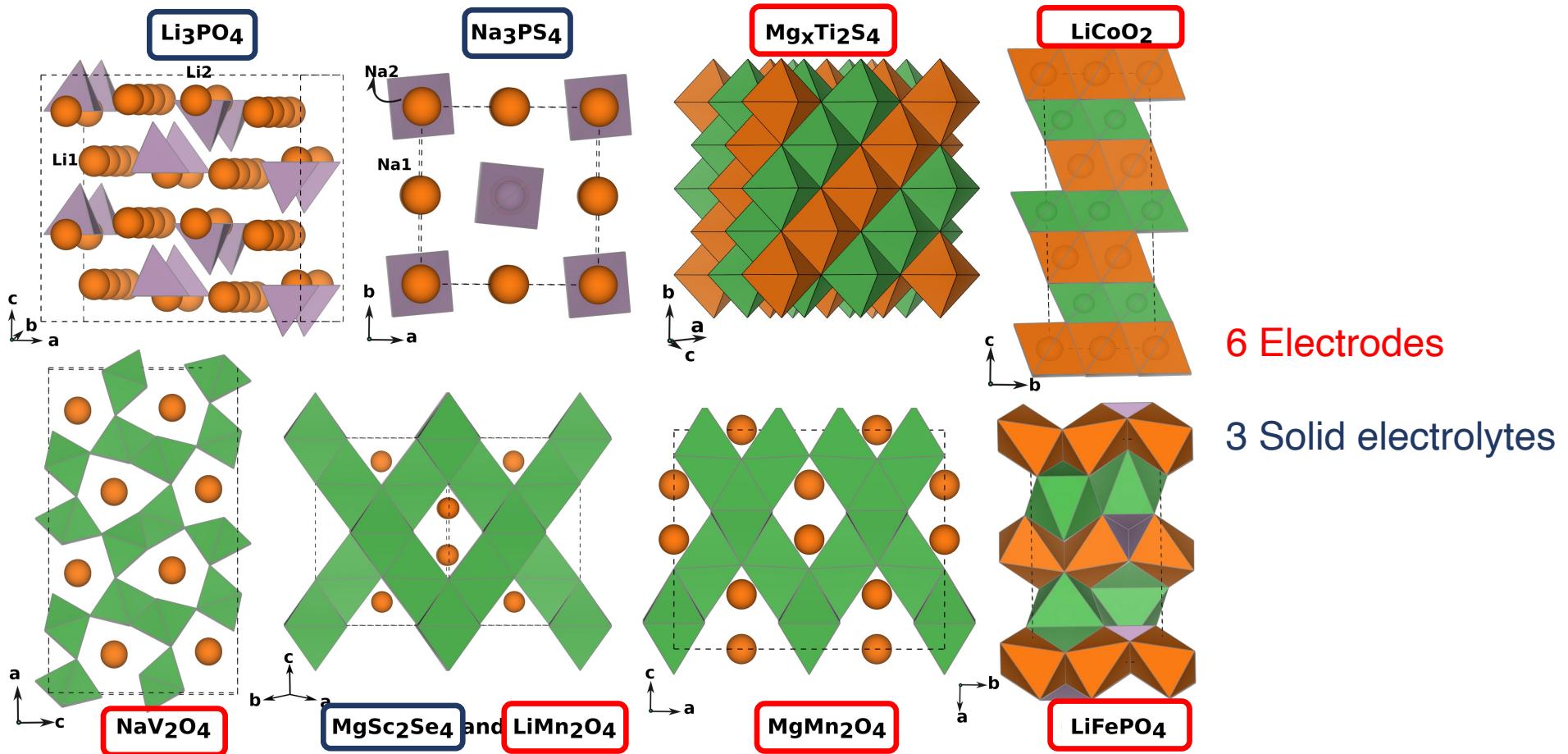
Hubbard U corrected frameworks³: GGA+ U , SCAN+ U
• For electrodes only

1. J.P. Perdew et al., Phys. Rev. Lett. 77, 3865 (1996)

2. J. Sun et al., Phys. Rev. Lett. 115, 036402 (2015)

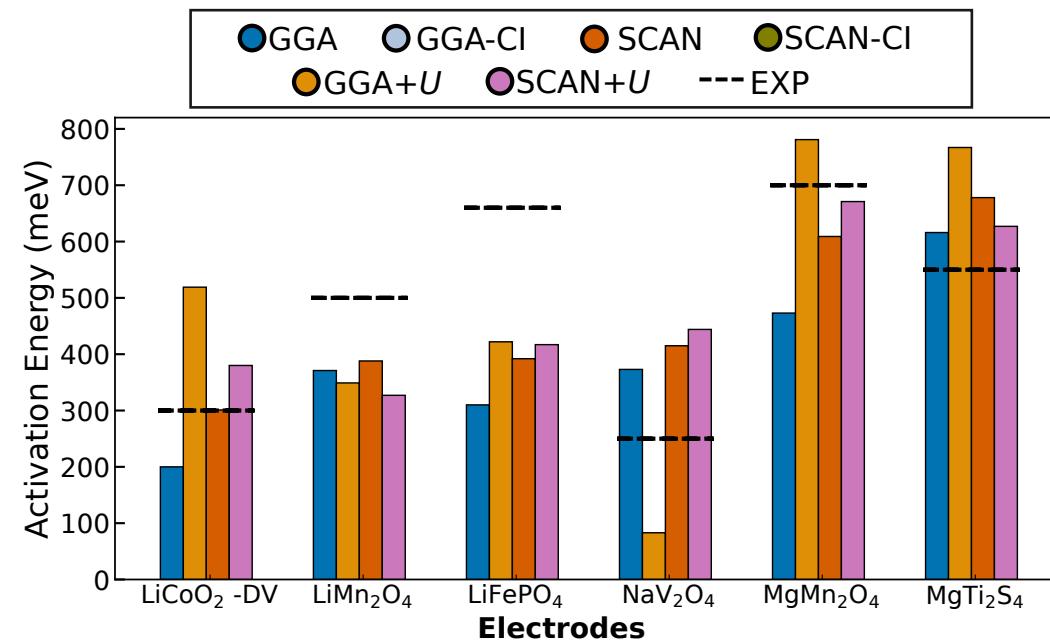
3. V. Anisimov et al., Phys. Rev. B 44, 943 (1991)

9 distinct systems considered



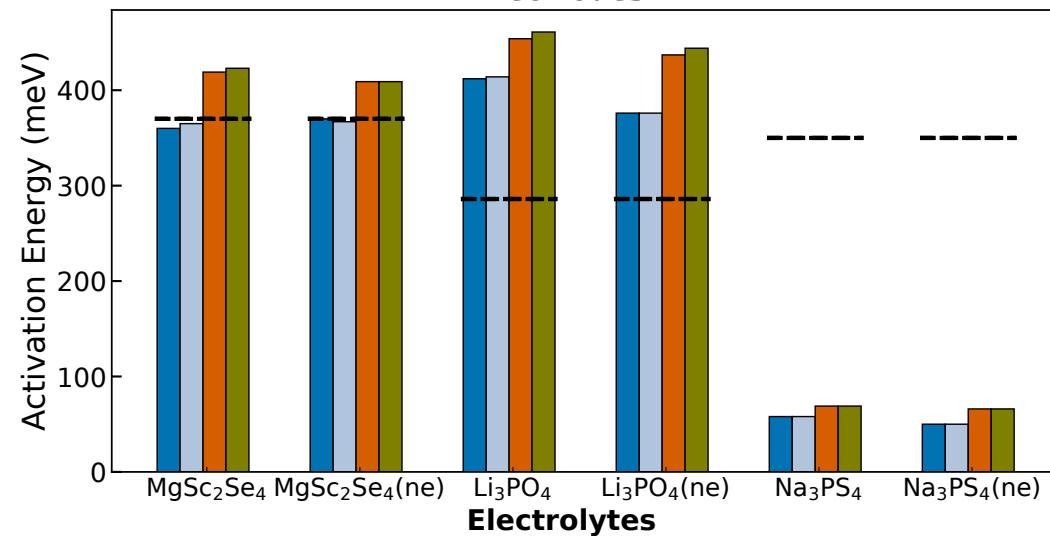
- Why these systems?
- Availability of experimental data
 - Heterogeneity of intercalation ion
 - Diversity of structural frameworks

SCAN exhibits better numerical accuracy, on average



MAE of SCAN (140 meV) lower than other XC frameworks (>145 meV)

SCAN $E_m >$ GGA E_m

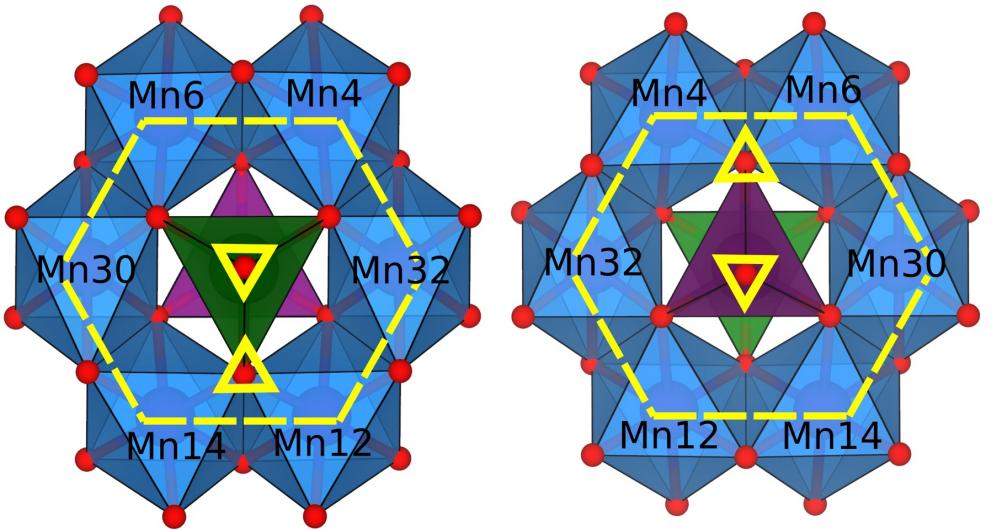


SCAN+U $E_m <$ GGA+U E_m

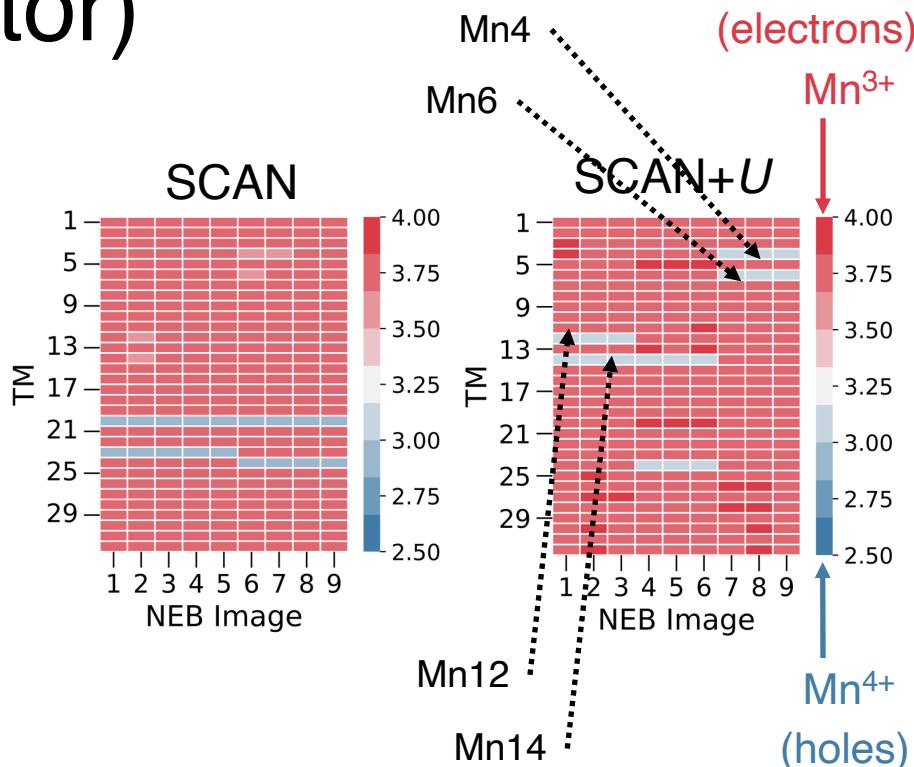
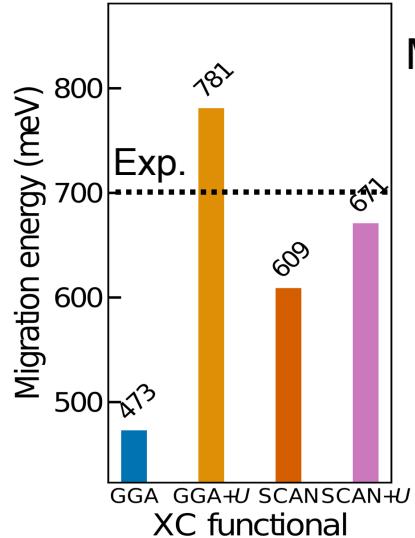
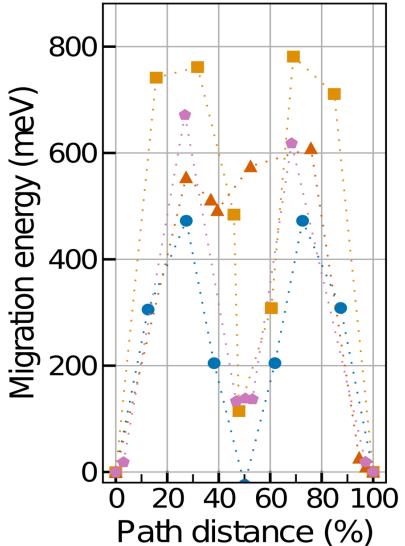
Addition of ne or CI doesn't seem to matter for solid electrolytes

Let's look at some specific cases

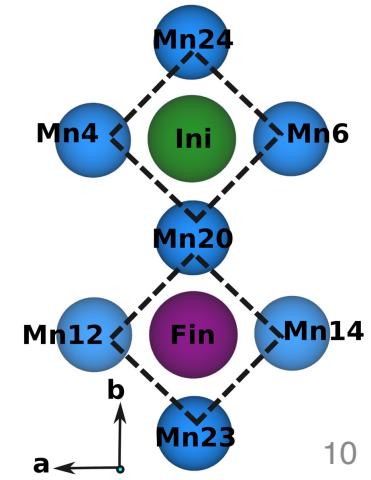
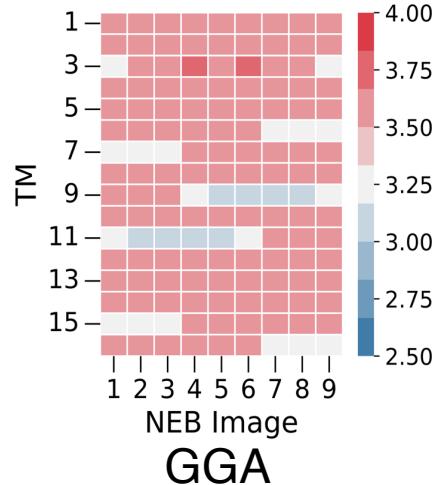
MgMn_2O_4 (semiconductor) GGA underestimates



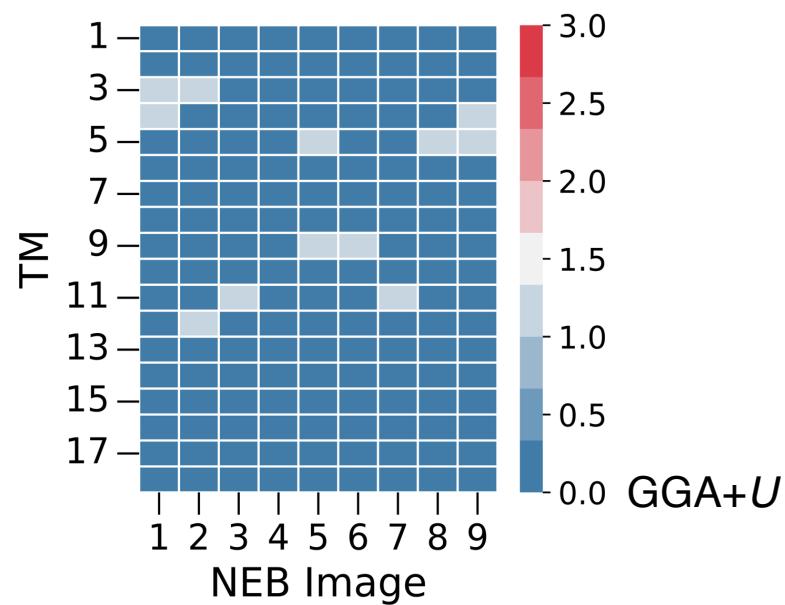
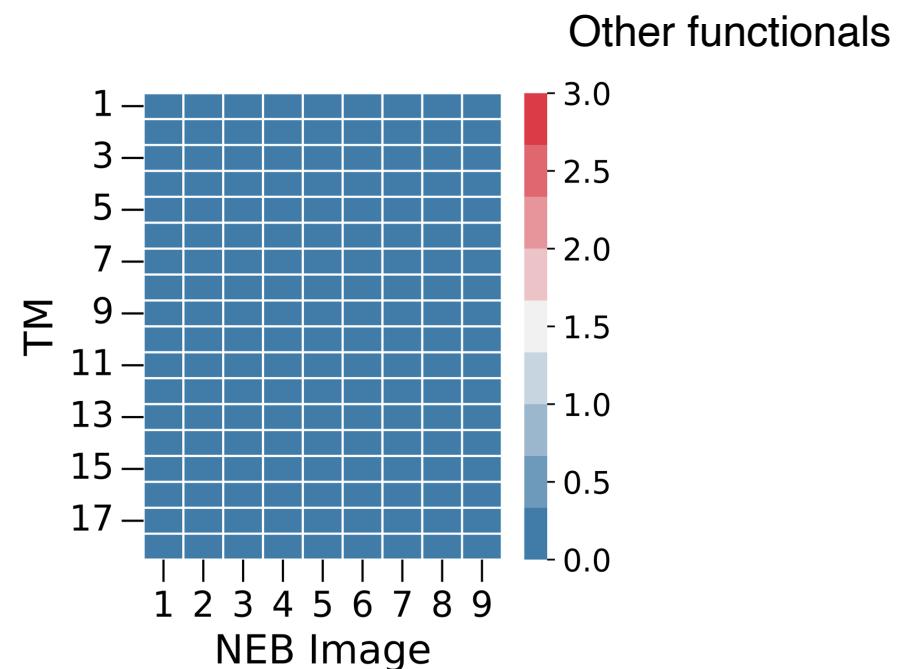
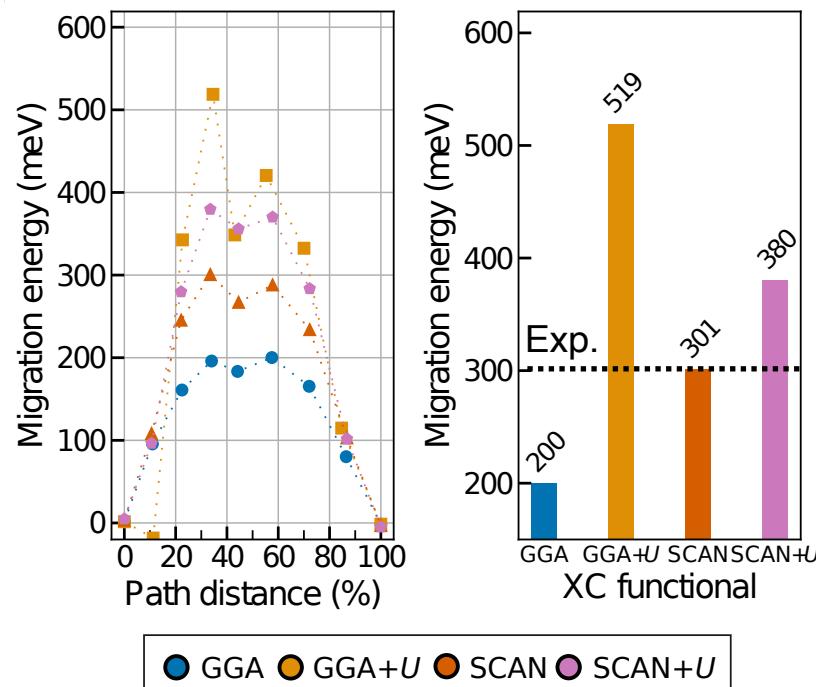
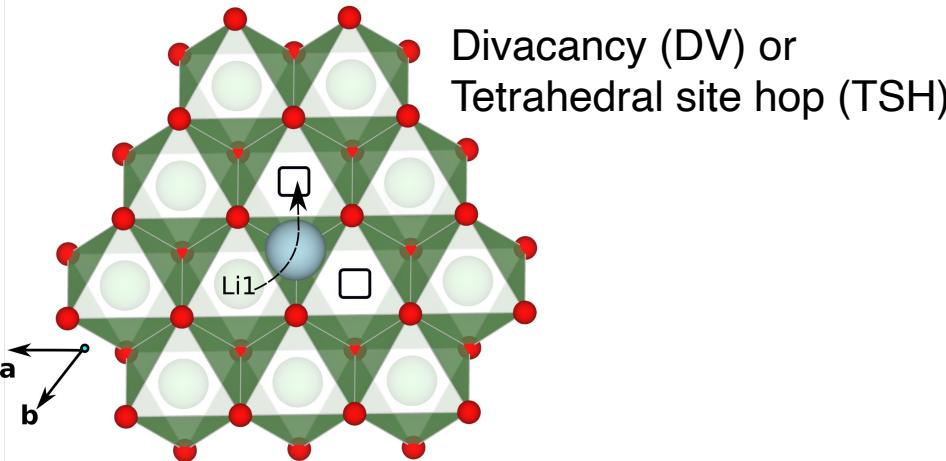
● GGA ● GGA+U ● SCAN ● SCAN+U



Mg^{2+} and holes should migrate in opposite directions

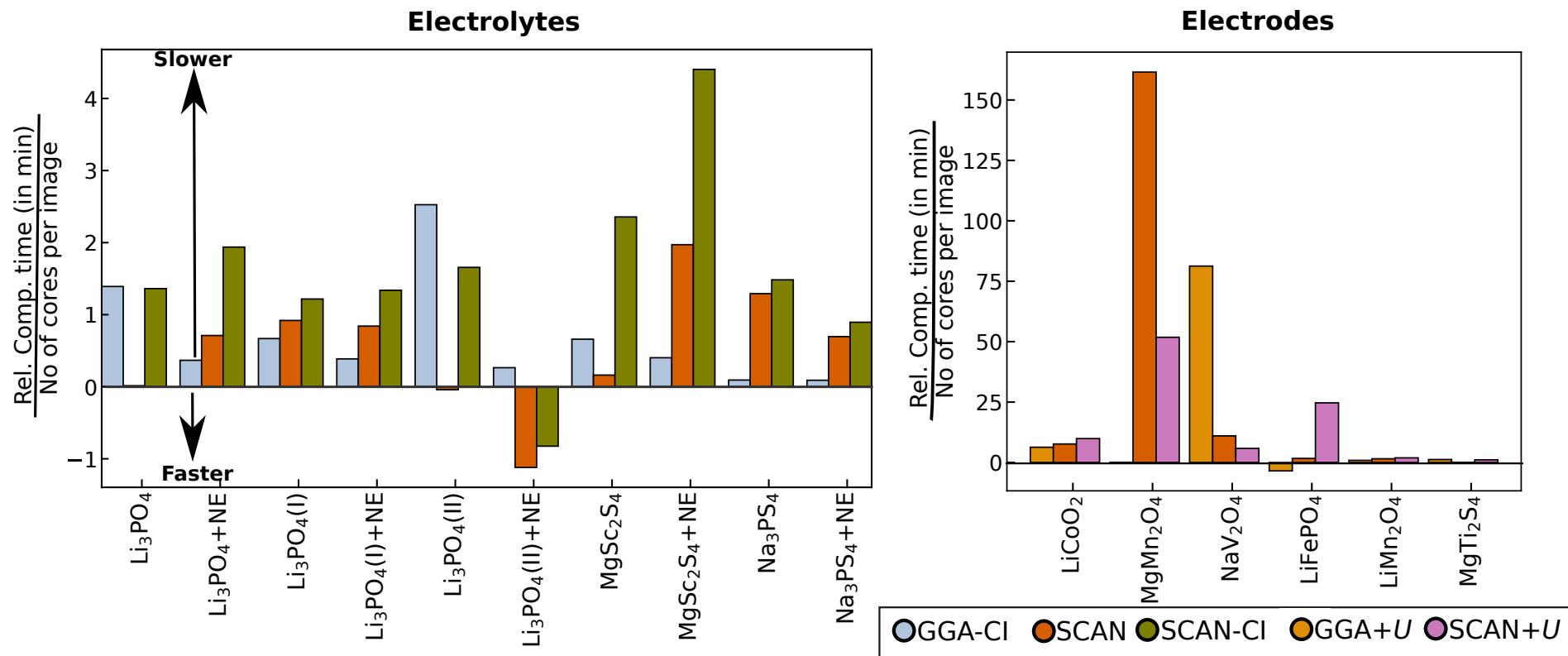


LiCoO_2 (metallic) GGA+ U overestimates



Computational performance

Is SCAN worth it?

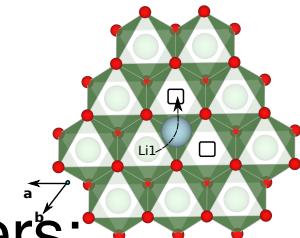


- Computational time with GGA/GGA+*U* ~75% faster than SCAN
- SCAN faster than SCAN+*U*
- Significant convergence difficulties with SCAN and SCAN+*U*

GGA for "quicker" estimate

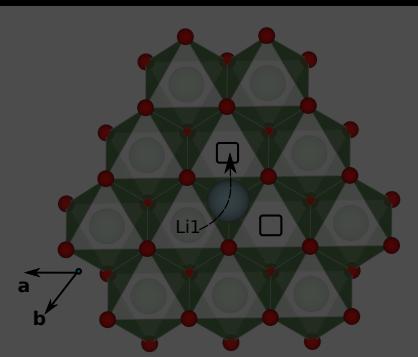
SCAN for "better" accuracy

Summary

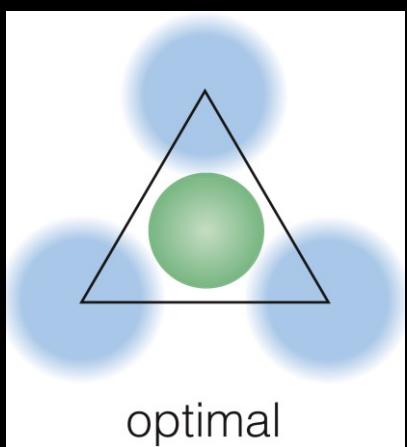


- Accurate computational predictions of migration barriers: important to improve rate performance of batteries
- NEB calculations – three computational handles relevant
 - XC framework
 - Uniform background charge
 - Climbing image
- Benchmark computational predictions with experimental barriers for 6 electrodes and 3 solid electrolytes
- SCAN: better accuracy on average
 - Describes better physics in both semiconducting and metallic hosts
 - Cl/ne doesn't influence migration barriers significantly
- Computational hosts+ convergence difficulties: high for SCAN
 - SCAN for better accuracy, but GGA can provide good qualitative trends

Objectives

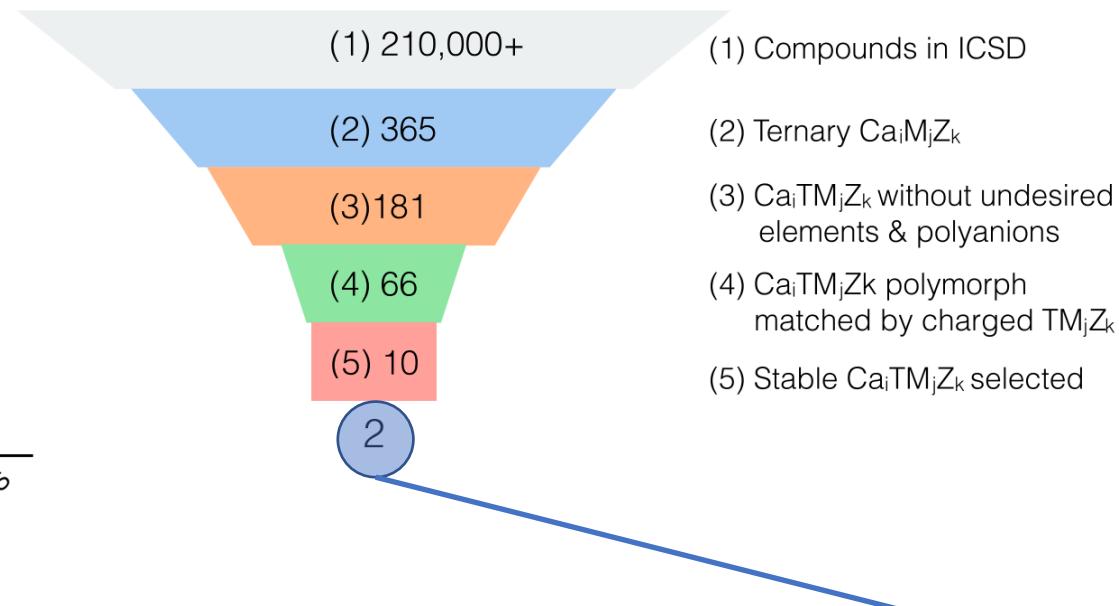
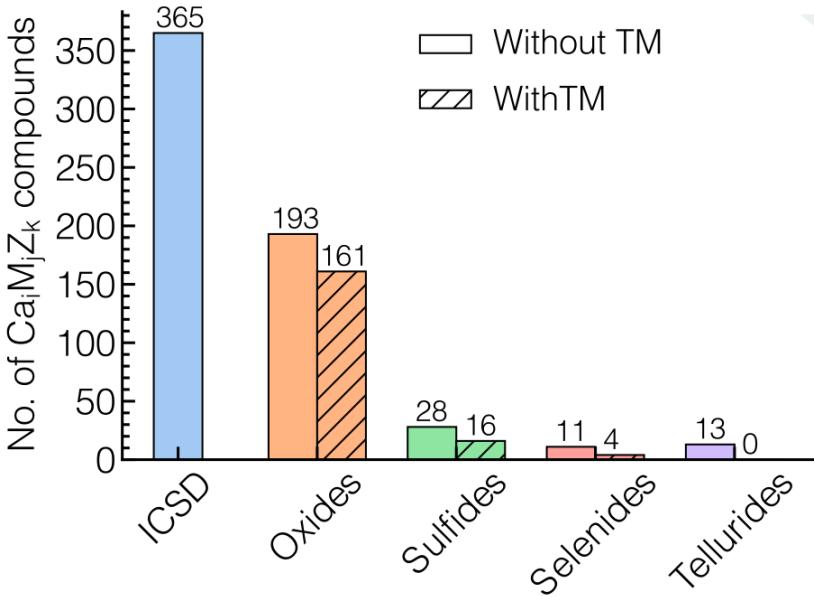


Assessing accuracy of computational predictions of ionic mobility



Examples of cathode discovery in Ca-batteries
(Ternary chemical space, NaSICONs)

Ternary Ca-compounds



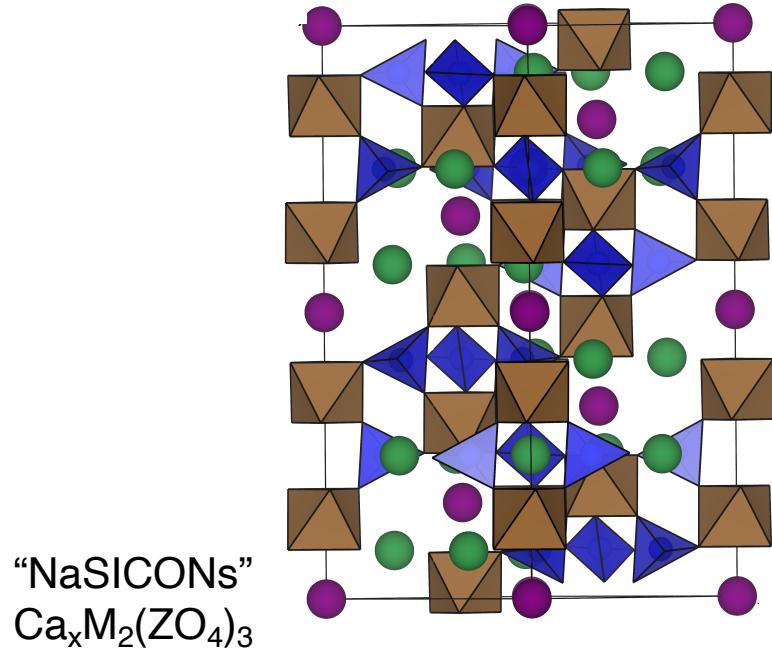
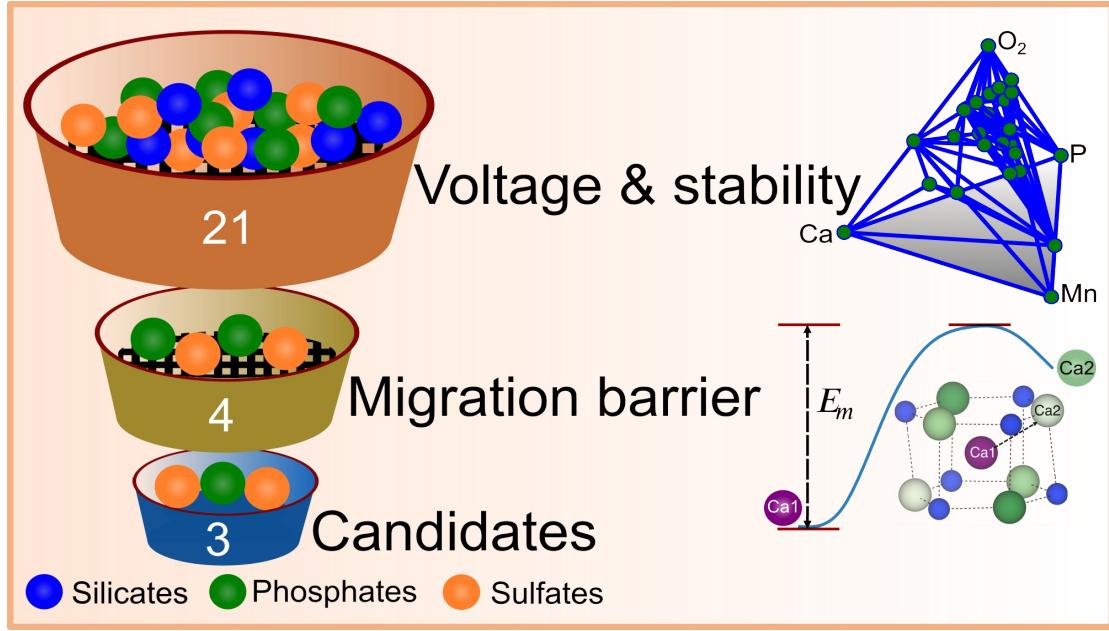
Inorganic crystal structure database (ICSD¹): has **> 210,000** compounds

- Only **365** are ternary compounds containing Ca
 - Compounds of composition $\text{Ca}_i\text{M}_j\text{Z}_k$; M, Z = elements other than Ca
- Let M = TM (i.e., transition metal) and Z = O, S, Se, or Te: **181** compounds
- Charge-neutral charged compound (TM_jZ_k) available for $\text{Ca}_i\text{TM}_j\text{Z}_k$?
 - CaMn_2O_4 - Mn_2O_4 is ok, CaVO_3 - VO_3 not ok
 - 66** unique structures
- Either of $\text{Ca}_i\text{TM}_j\text{Z}_k$ or TM_jZ_k thermodynamically (meta)stable?
 - $E^{\text{hull}} \leq 30$ meV/atom (based on Materials Project²)
 - 10** unique compounds → evaluate (voltage and) mobility

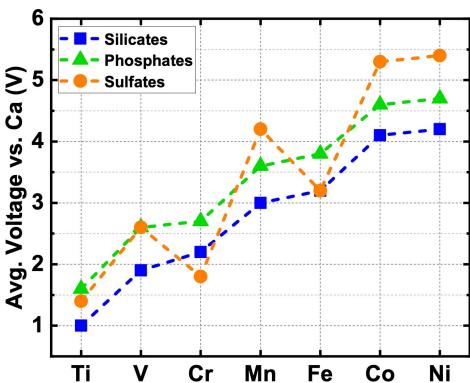
Final candidates!
 CaV_2O_4
and
 CaNb_2O_4

1. <https://icsd.products.fiz-karlsruhe.de/>
2. <https://materialsproject.org/>

Sodium superionic conductors (NaSICONs)

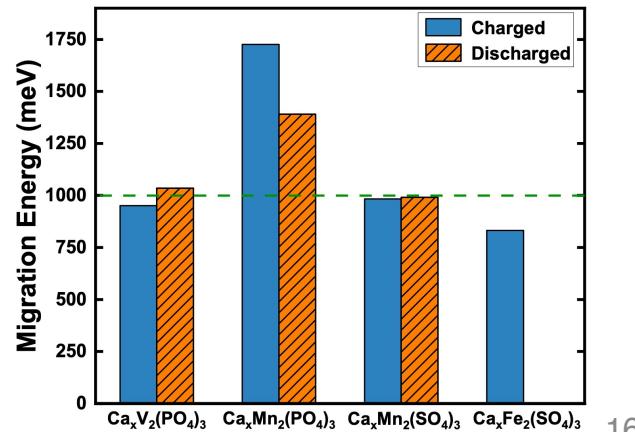


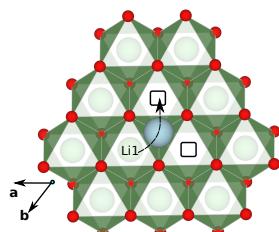
High-throughput DFT calculations: 3 candidates



$\text{Ca}_x\text{V}_2(\text{PO}_4)_3$, $\text{Ca}_x\text{Mn}_2(\text{SO}_4)_3$, and $\text{Ca}_x\text{Fe}_2(\text{SO}_4)_3$

	Ti	V	Cr	Mn	Fe	Co	Ni
$\text{Ca}_2\text{M}_2(\text{SiO}_4)_3$	71	93	706	111	192	237	269
$\text{Ca}_4\text{M}_2(\text{SiO}_4)_3$	93	100	450	83	93	84	110
$\text{Ca}_{0.5}\text{M}_2(\text{PO}_4)_3$	-45	-8	12	-23	92	194	1173
$\text{Ca}_{2.5}\text{M}_2(\text{PO}_4)_3$	129	54	108	-11	35	50	693
$\text{M}_2(\text{SO}_4)_3$	-159	-107	-224	-74	-182	64	71
$\text{CaM}_2(\text{SO}_4)_3$	174	63	172	21	29	27	27





Conclusions

- Removing material bottlenecks is important for improving performance of energy devices
 - Need cathodes and solid electrolytes with good power performance
 - Need robust computational predictions of ionic mobility
- SCAN provides better accuracy on E_m predictions, on average
 - Computational costs + convergence difficulties: high
- Ca-containing ternary compounds and NaSICONs screened
 - 2 possible candidates: CaV_2O_4 and CaNb_2O_4
 - $\text{Ca}_x\text{V}_2(\text{PO}_4)_3$, $\text{Ca}_x\text{Mn}_2(\text{SO}_4)_3$, and $\text{Ca}_x\text{Fe}_2(\text{SO}_4)_3$ are promising

NEB Benchmarking:

“Effect of exchange-correlation functionals on the estimation of migration barriers in battery materials”, R. Devi, B. Singh, P. Canepa, and G. Sai Gautam, **npj Comput. Mater.** **2022**, *8*, 160

Ternary Ca-cathode screening:

“Searching ternary oxides and chalcogenides as positive electrodes for calcium batteries”, W. Lu, J. Wang, G.Sai Gautam, and P. Canepa, **Chem. Mater.** **2021**, *33*, 5809-5821

Ca-NaSICON screening:

“Exploration of NaSICON frameworks as calcium-ion battery cathodes”, D.B. Tekliye, A. Kumar, X. Weihang, T.D. Mercy, P. Canepa, and G.Sai Gautam, **Chem. Mater.** **2022**, *34*, 10133-10143