

Optimal use of transfer learning for predicting ionic migration barriers in solids

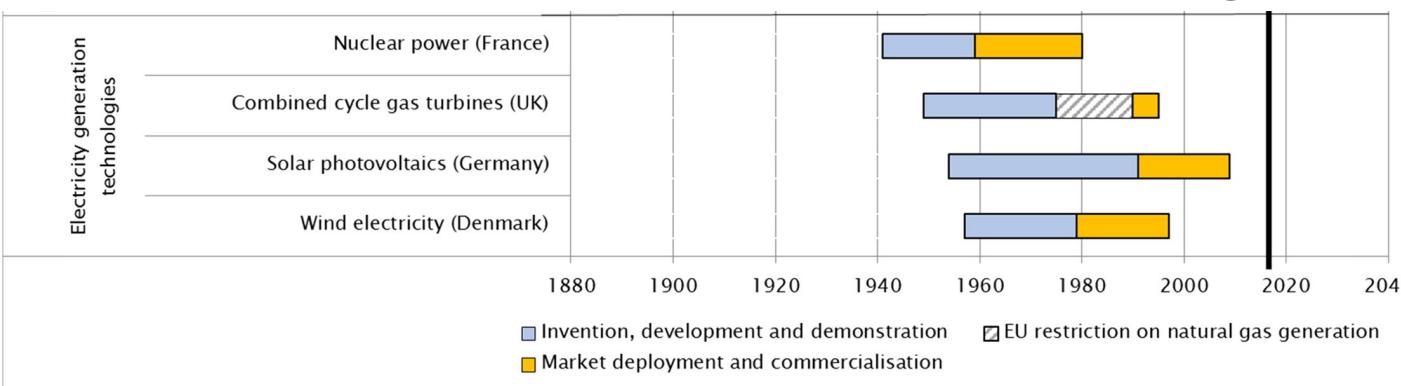
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We work broadly on energy materials

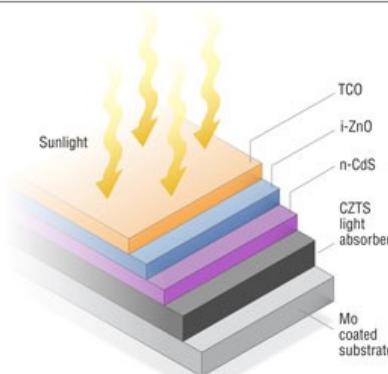


Technological innovation cycle → slow in energy sector

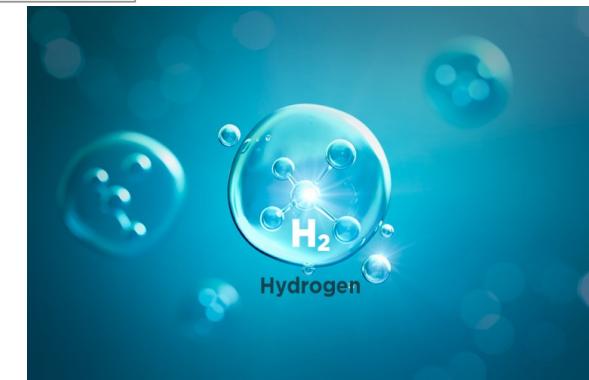
Need rapid identification of novel materials for improving performance



Design better electrodes and solid electrolytes



Develop better light-absorbing semiconductors

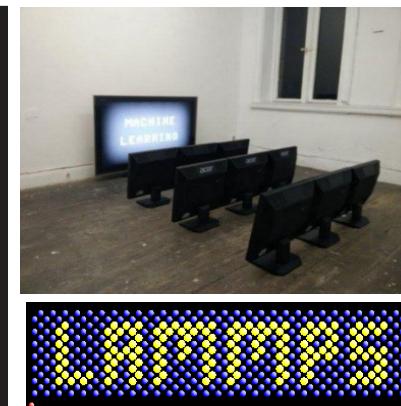


Identify better materials for H_2 generation and sensing

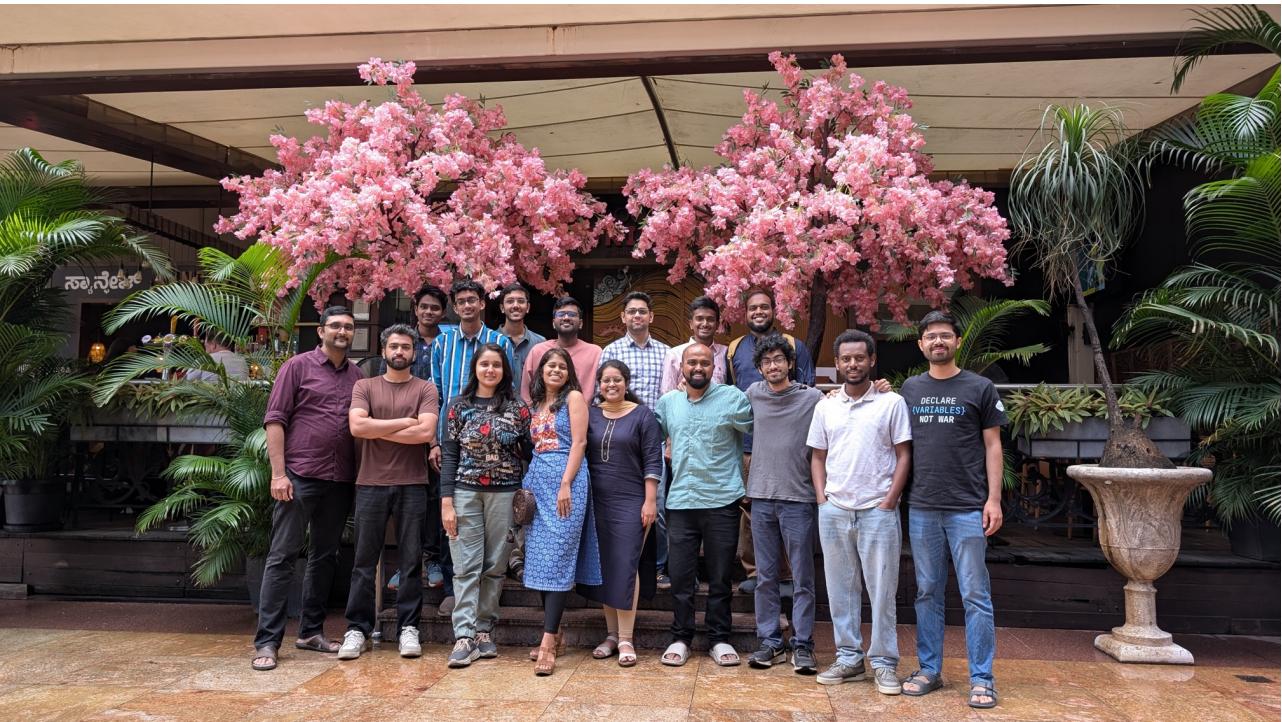
Identify novel materials for applications

Understand underlying materials phenomena better

Make theory better/build models



Acknowledgments



Group picture in May 2025



Param
Utkarsh
(CDAC)



Archer
(UK)



Dr. Keith Butler



Reshma



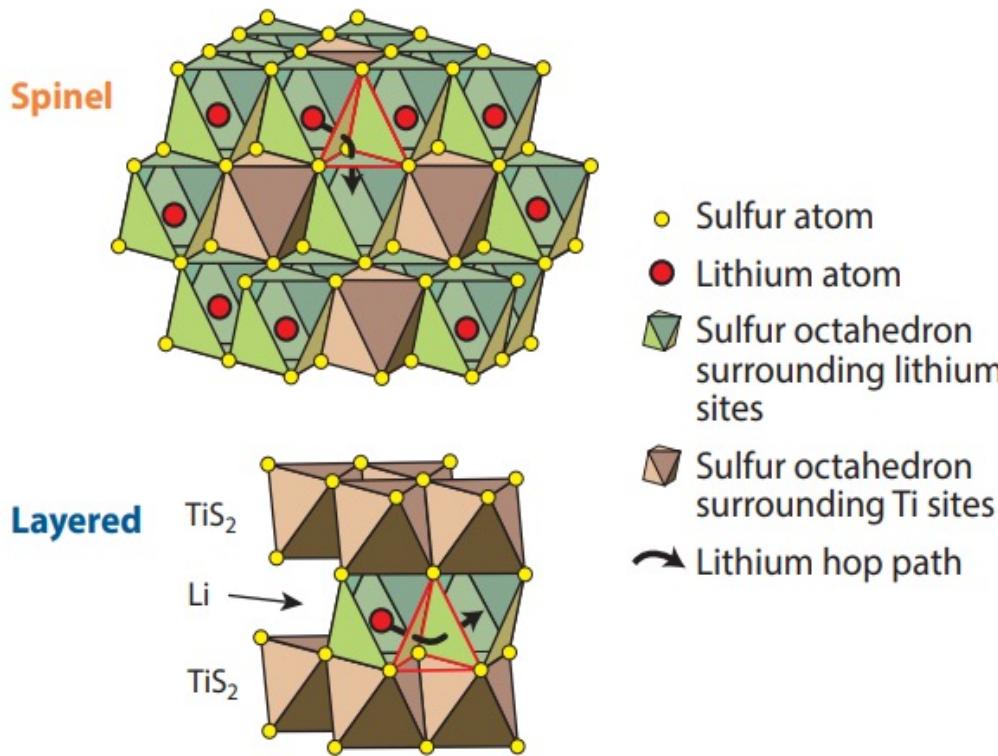
Avaneesh



SERC (IISc)



Migration barriers govern rate performance in batteries



Intercalation electrodes: ionic diffusivity (D) within the bulk a major factor in rate performance

$$D = D_o \exp\left(-\frac{E_m}{k_B T}\right)$$

D_o : Diffusivity pre-factor (carrier concentration, correlations, etc.)

k_B : Boltzmann constant

T : Temperature

E_m : Migration barrier

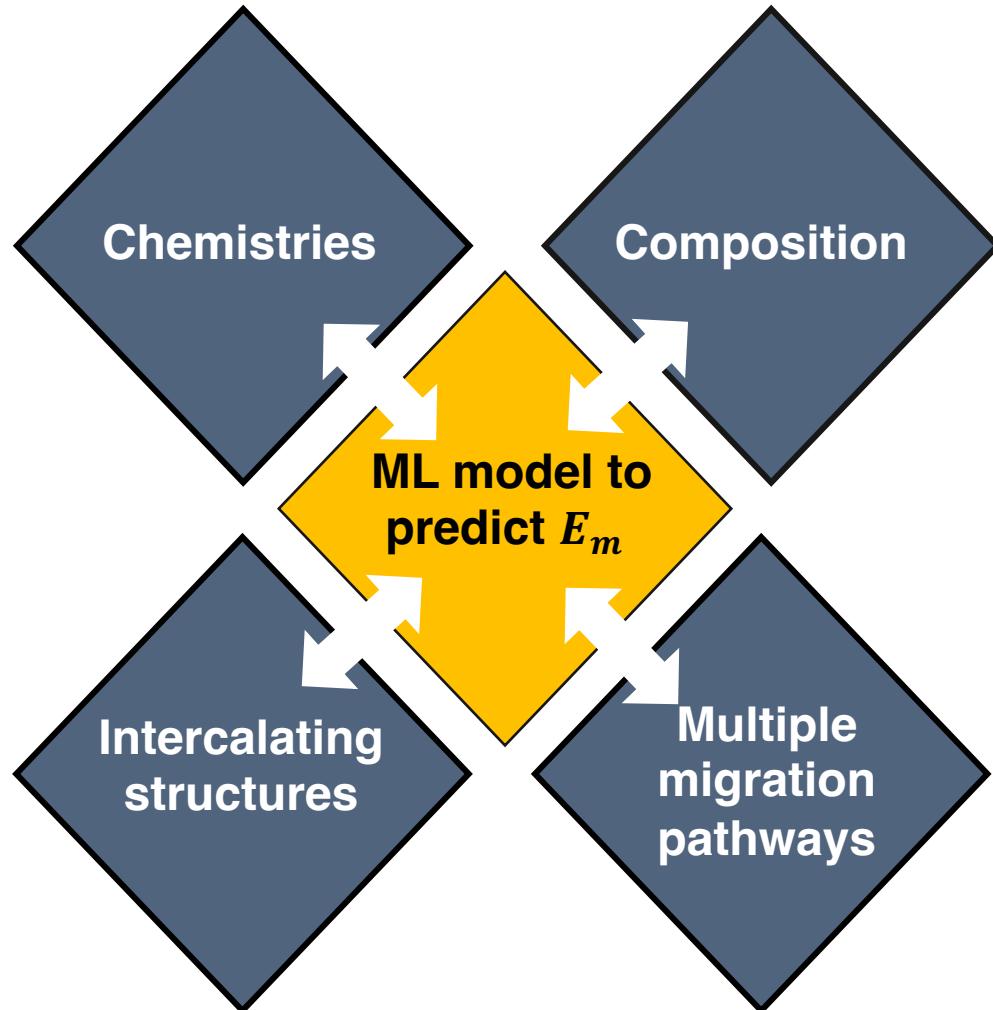
Migration barrier: dominant factor determining diffusivity

Experimental measurements of E_m : variable-temperature impedance spectroscopy, variable temperature nuclear magnetic resonance, etc.

Computational predictions of E_m : ab initio molecular dynamics, nudged elastic band (NEB) with density functional theory (DFT)

How to accurately and swiftly predict migration barriers? Use machine learning?

What we need for fast and accurate E_m predictions?



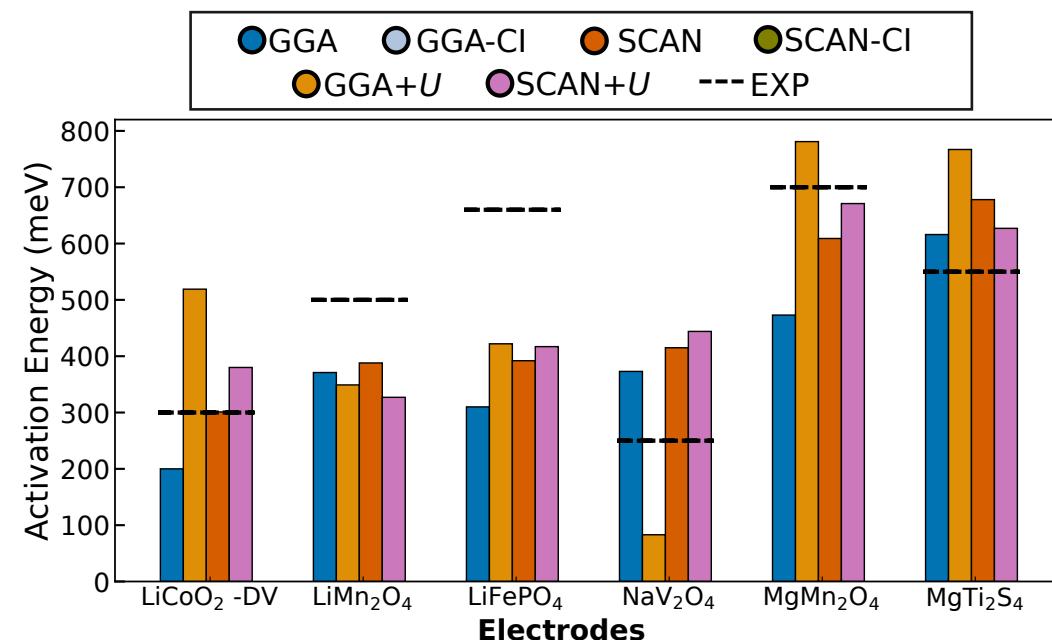
How **accurate** are first-principles calculations in predicting E_m ?

Is there a **dataset** of E_m values belonging to a diverse set of structures, chemistries, compositions, and migration pathways?

If there is a dataset, how do we construct **a generalizable model** that accurately predicts E_m ?
• Using transfer learning

Accuracy

On average accuracy is in the 140-180 meV range



MAE of SCAN (140 meV): lowest

MAE of GGA: 178 meV

Qualitative trends: reliable with GGA

SCAN: computational cost+ convergence

60 meV in E_m : ~1 order of magnitude D at 300 K for micron-sized particles

125 meV in E_m : same D at 300 K, one order change in particle size

SCAN: strongly constrained and appropriately normed

GGA: Generalized gradient approximation

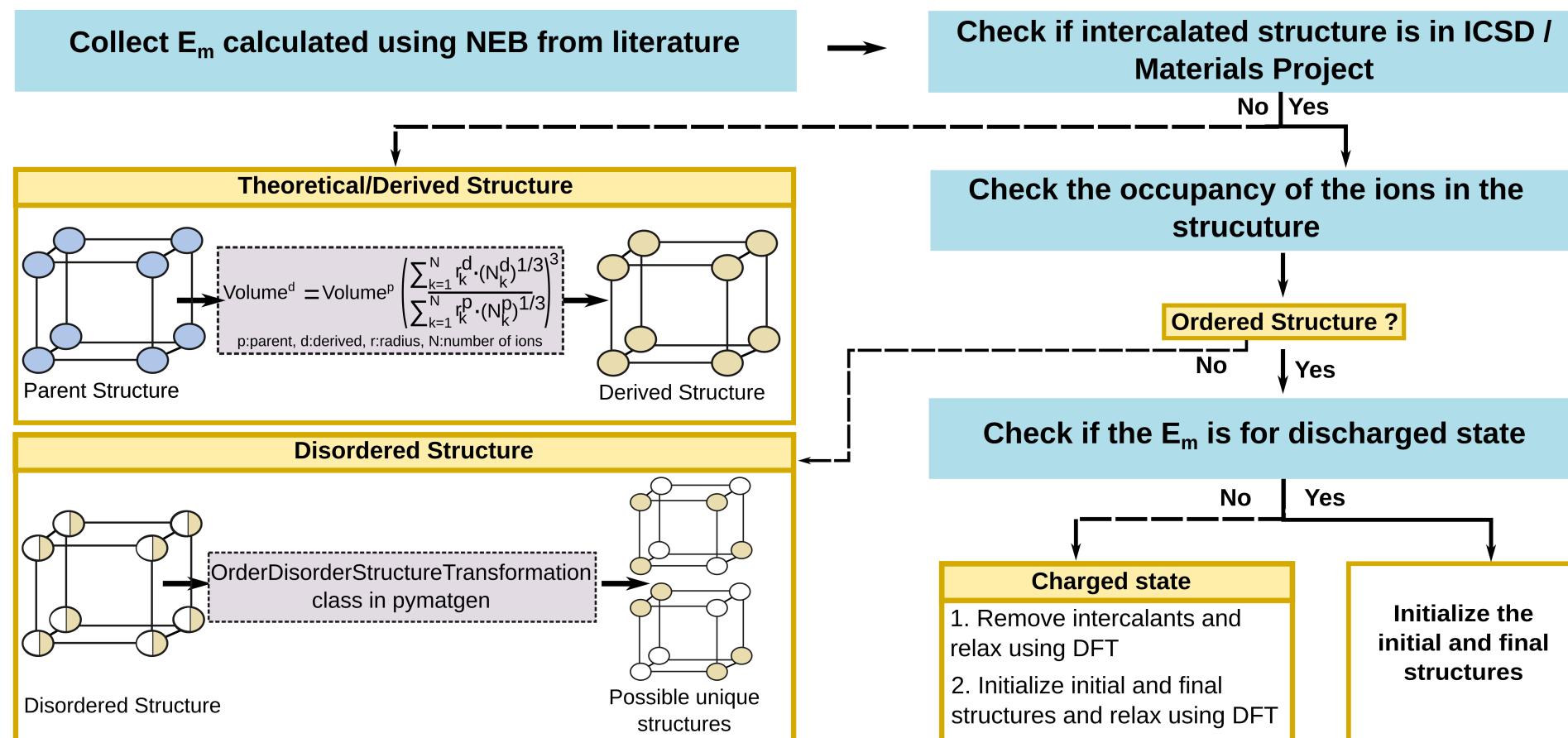
U : Hubbard U correction

CI: Climbing image; EXP: Experimental

ne: Background charge; DV: divacancy

Dataset

Workflow to generate a curated dataset of E_m



- Consider papers that reported unambiguously the methods and the structure used
- Generalized gradient approximation or its Hubbard U preferred as the functional
- Well labelled images of E_m and/or the minimum energy pathway (MEP)

Statistics

619 datapoints spanning
58 space groups across
7 crystal systems

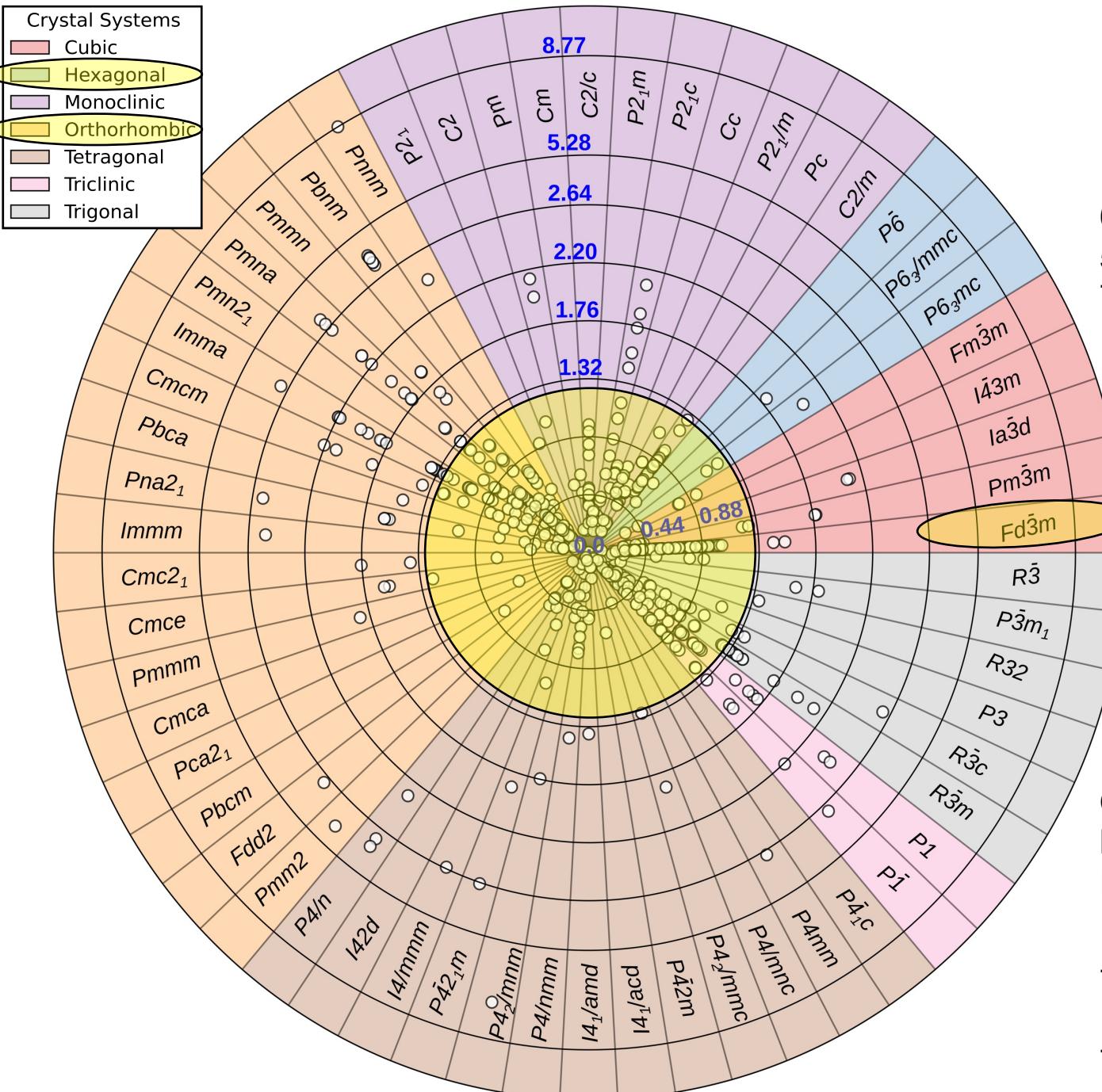
Range: 0.03 to 8.77 eV

528 electrodes
91 electrolytes

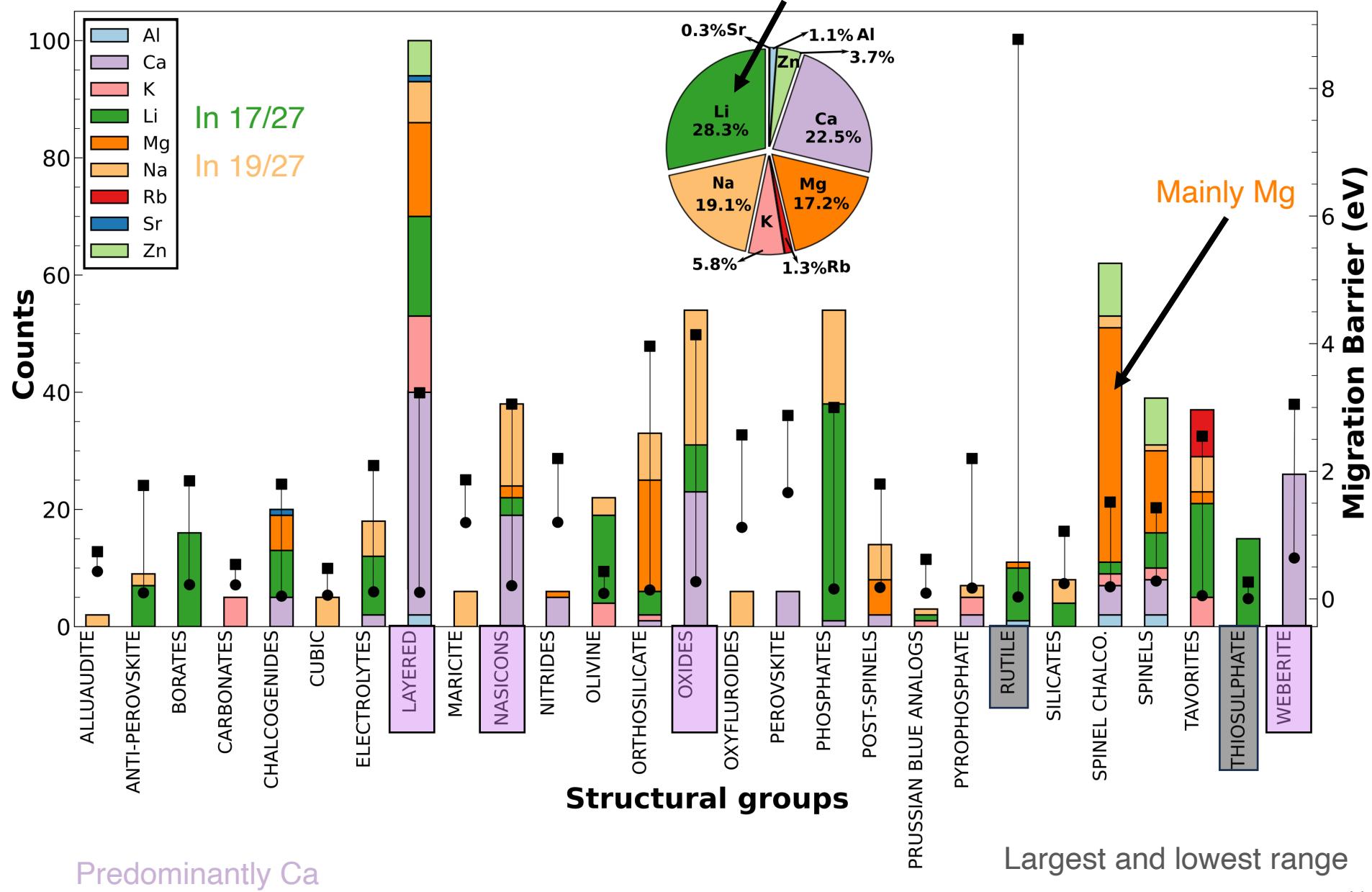
Cubic spinels: 94 entries

Orthorhombic (206):
highest
Hexagonal (6): lowest

73.4%: $E_m < 1$ eV
19.4%: $1 \text{ eV} < E_m < 2$ eV
7.2%: $E_m > 2$ eV



Data distribution

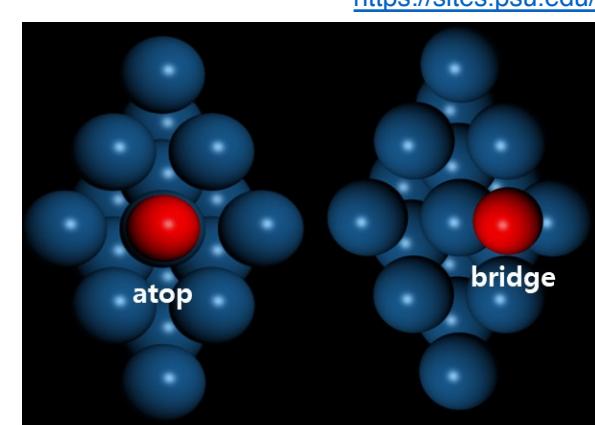
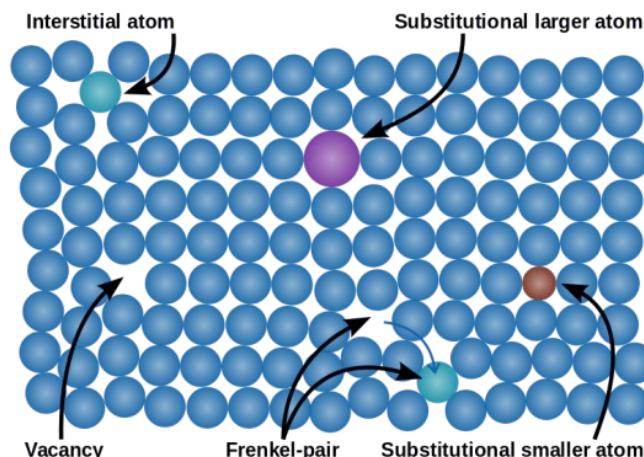
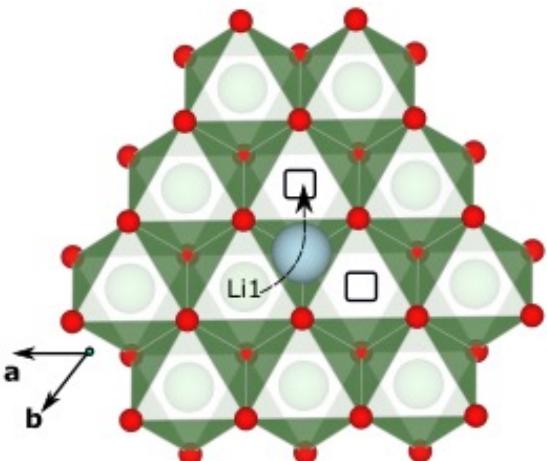


How to create generalizable models? Transfer learning

Materials science is data limited

Several key material properties that govern performance in applications have limited data

- ‘Small’ datasets ($< 10^4$ datapoints)
 - Ionic mobilities, defect formation energies, adsorption energies,...
- Limits application of deep learning (DL) frameworks



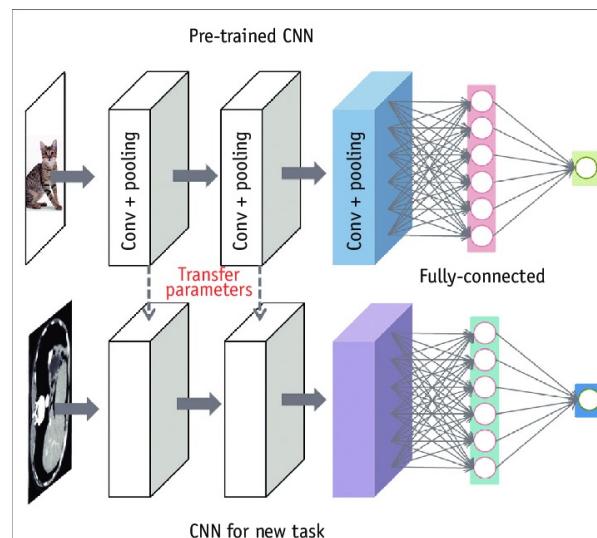
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Transfer learning: efficiently use DL frameworks on small datasets

- Pre-train (**PT**) on ‘large’ dataset, fine-tune (**FT**) on ‘small’ dataset



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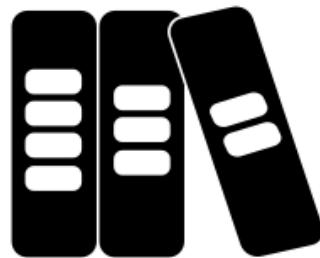
Transfer learning: efficiently use DL frameworks on small datasets

- Pre-train (**PT**) on ‘large’ dataset, fine-tune (**FT**) on ‘small’ dataset

How useful is transfer learning in materials science?

- Optimal ways to use?
- Ways to generate ‘generalizable’ models?

Handles to consider

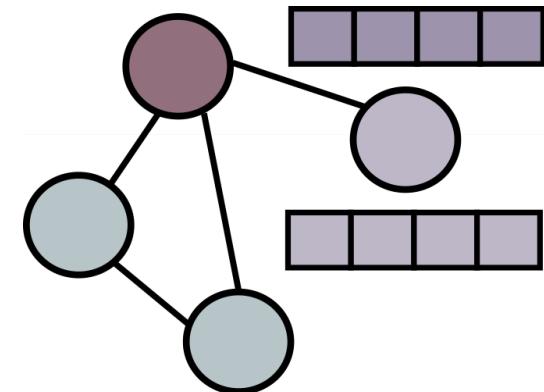


Dataset(s)

- What, how, how many?

Architecture

- Graph neural network



Frozen

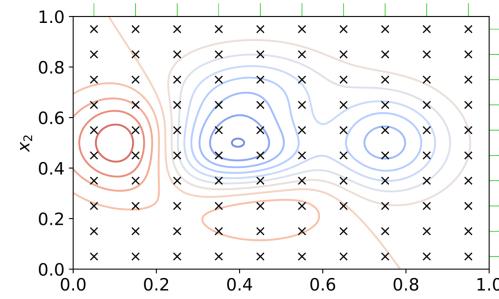
Unfrozen

Strategy

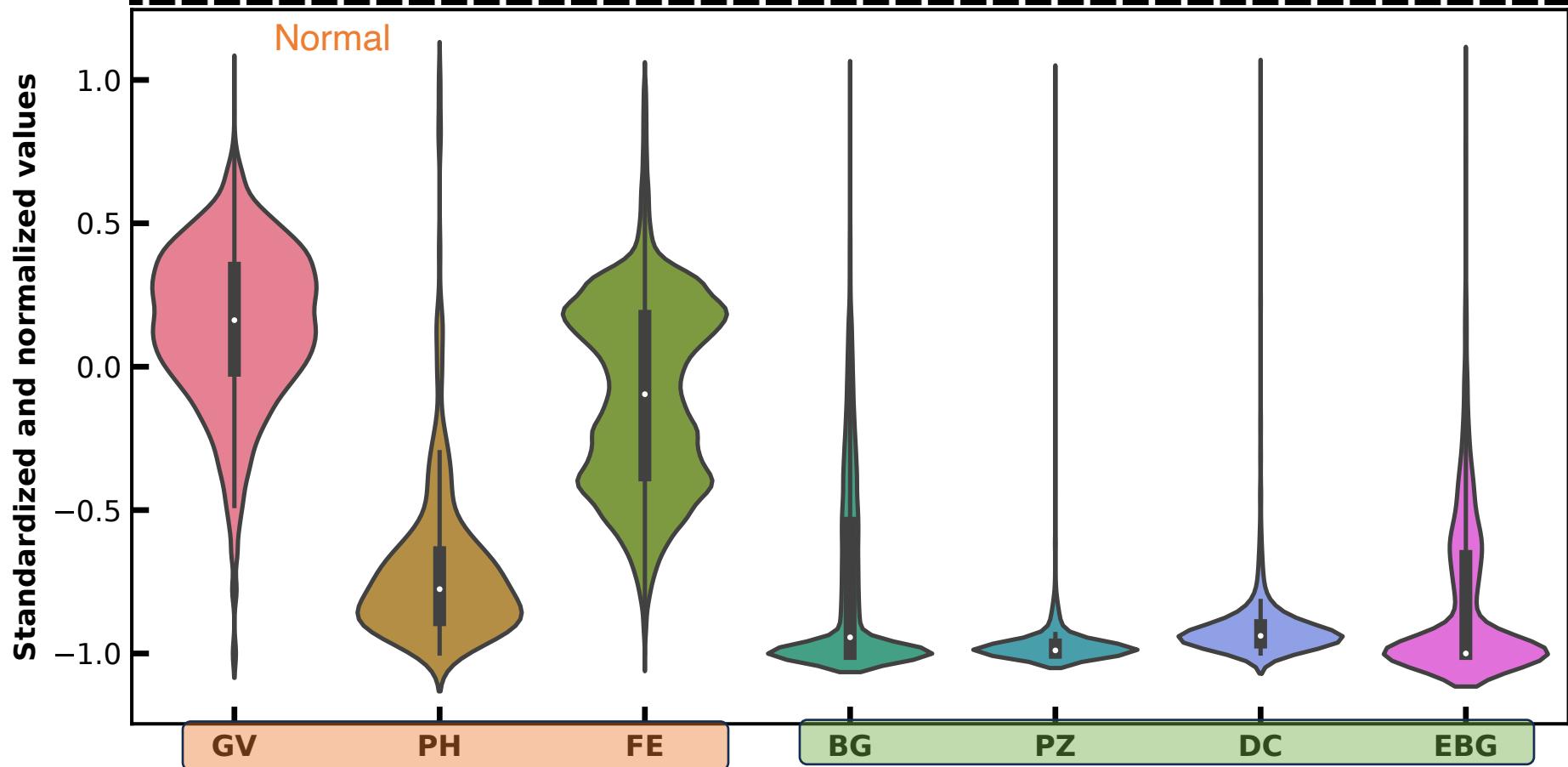
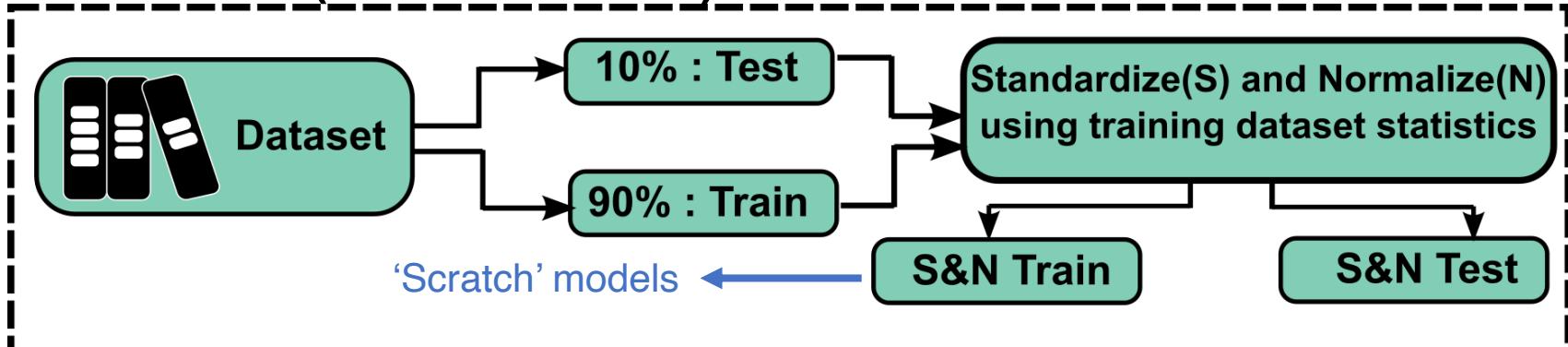
- FT techniques in pair-wise PT/FT models
- Multi-property PT (**MPT**) models

(Learning) Hyperparameters

- Data sampling
- Learning rate
- Number of datapoints during PT, FT

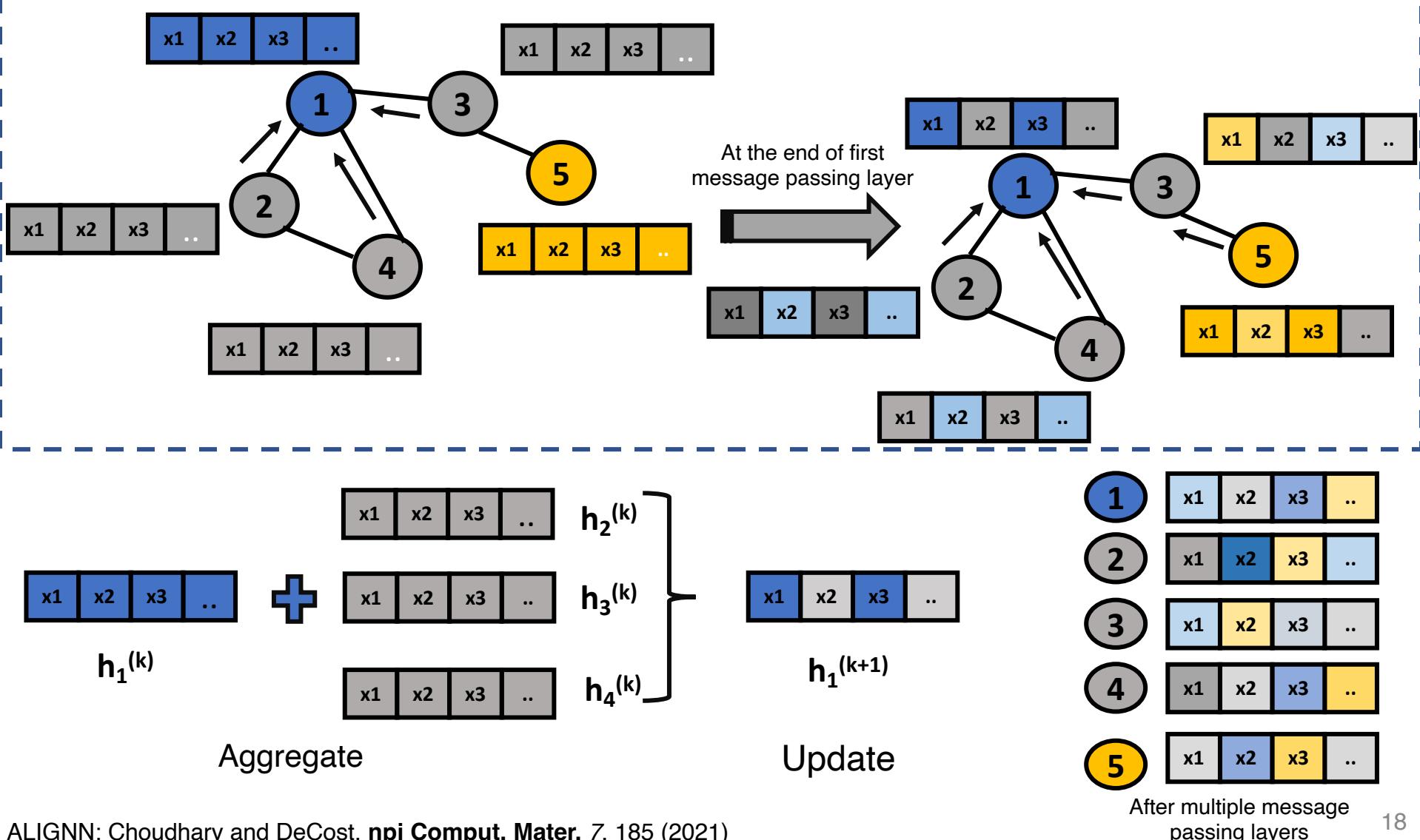


7 datasets (Matminer)

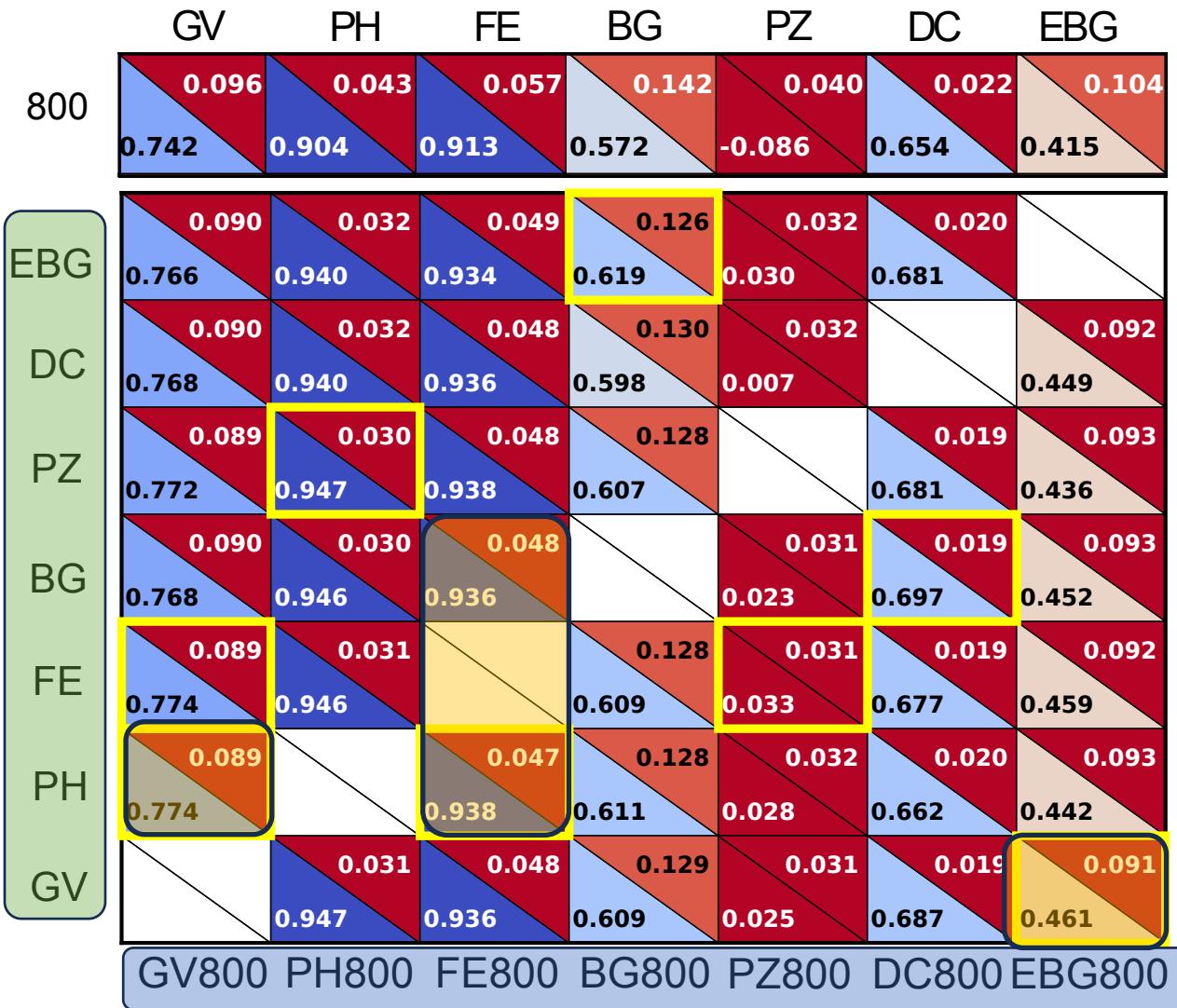


Base model architecture: graph neural networks

Specific architecture: 'ALIGNN'



7×6 combinations of pair-wise models



GV: Shear modulus; PH: Phonons; FE: Formation energy; BG: Band gap

PZ: Piezoelectric modulus; DC: Dielectric constant; EBG: Experimental band gap

Pair-wise models:
better than scratch

- Average increase in R^2 : 25%
- Average decrease in MAE: 16%

Best models: GV,
PH, FE ($R^2 > 0.75$)

Average models:
BG, DC, EBG

Specific PT
property: little
influence on FT

No symmetry

Test scores

MAE

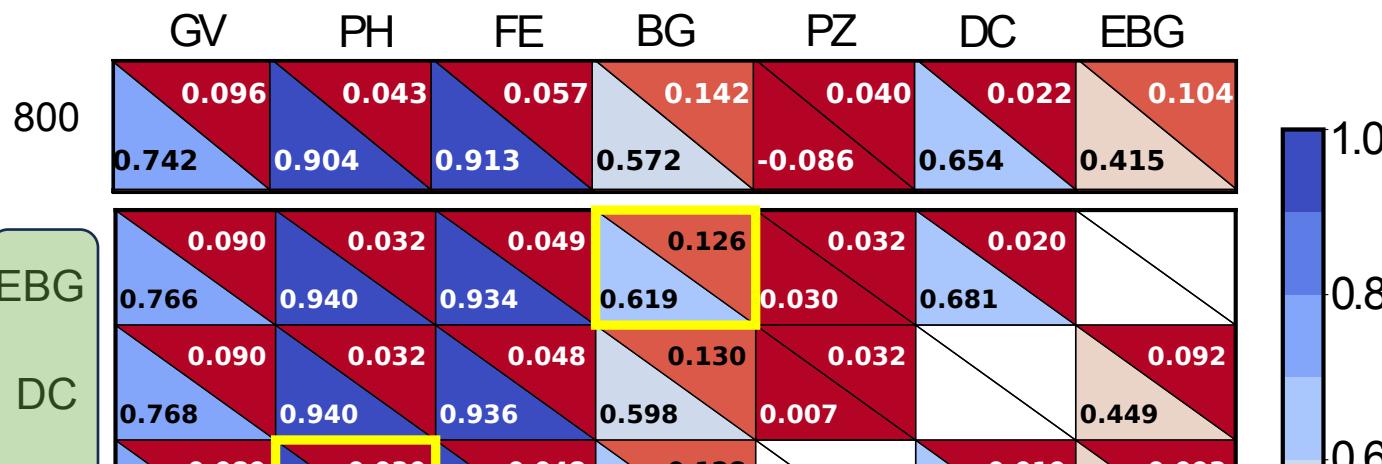
R^2

FT dataset+size

PT dataset (941)

Best model

7×6 combinations of pair-wise models



Pair-wise models:
better than scratch

- Average increase in R^2 : 25%
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Best models: GV,
PH, FE ($R^2 > 0.75$)

What about MPT (or more generalizable) models?



property: little influence on FT

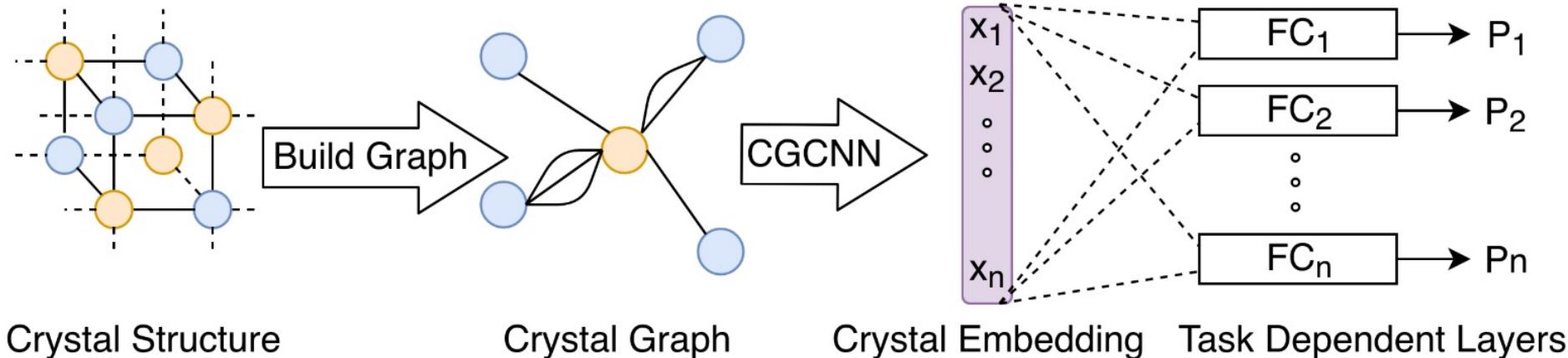
No symmetry

At capped dataset size, specific PT property is a weak handle; Normal distribution is better

Pair-wise transfer learning has significant utility

MPT: (Beta) Generalized models

Inspiration from literature: multi-task crystal graph convolutional neural network¹



Crystal Structure

Crystal Graph

Crystal Embedding

Task Dependent Layers

MPT models can generalize dependence of several properties on the structure

- Build cumulative dataset: 132,270 points
 - Remove overlaps
- Add task-dependent prediction heads with a one-hot encoded vector
 - Presence/absence of property
- Modify loss function
- PT on all (but one) property, FT on one property

$$\mathcal{L} = \frac{1}{N} \sum_{i=1}^N |y_p^i - y_t^i| \delta^i$$

MPT: better out-of-domain than PT-FT

Band gap of 2D materials (1,103 datapoints) from JARVIS-DFT¹

Model	Test R ²	Test MAE
Scratch	0.635	0.148
MPT (all seven datasets)	0.671	0.125
FE(100K)	0.670	0.127
BG(50K)	0.617	0.138
PH(1256)	0.628	0.145
GV(10,987)	0.626	0.143
EBG(2,481)	0.619	0.143

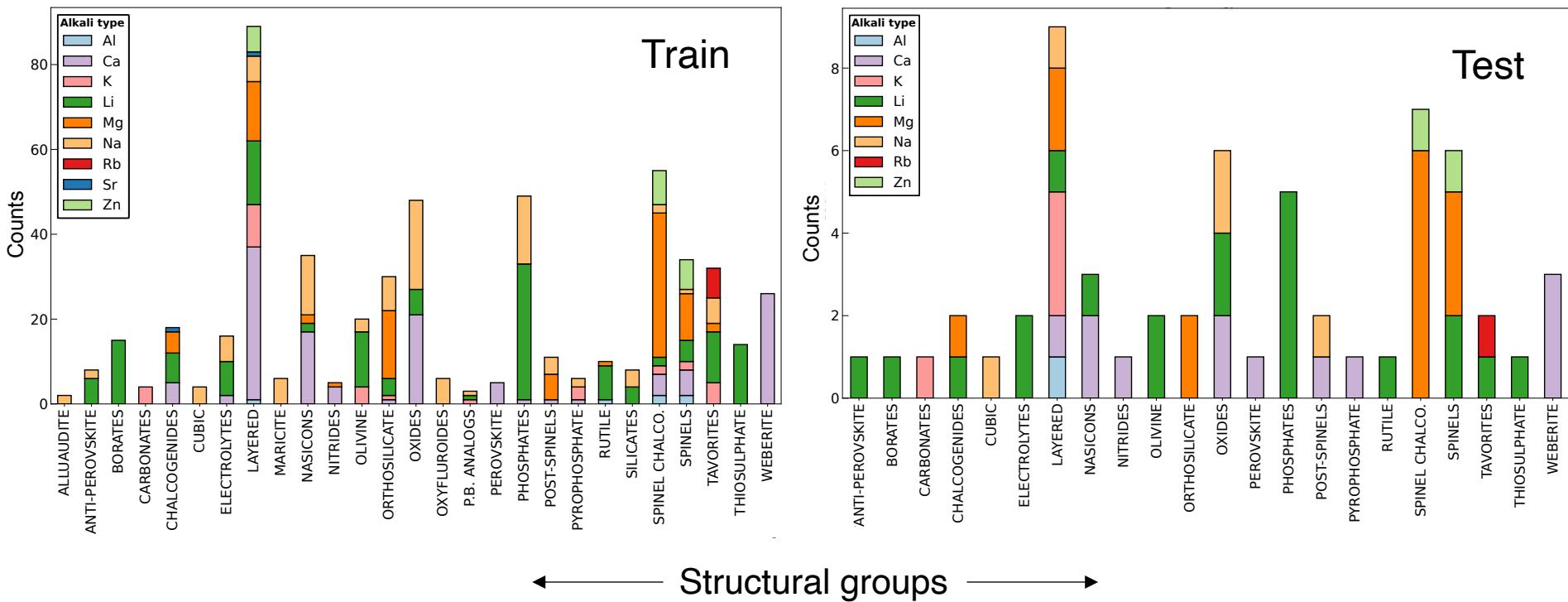
On average, MPT is 6% and 10% better on R² and MAE than PT-FT

Closest performer to MPT is FE: largest dataset within MPT

MPT models: may generalize quite well with more properties

How to predict migration barriers? Using the MPT model

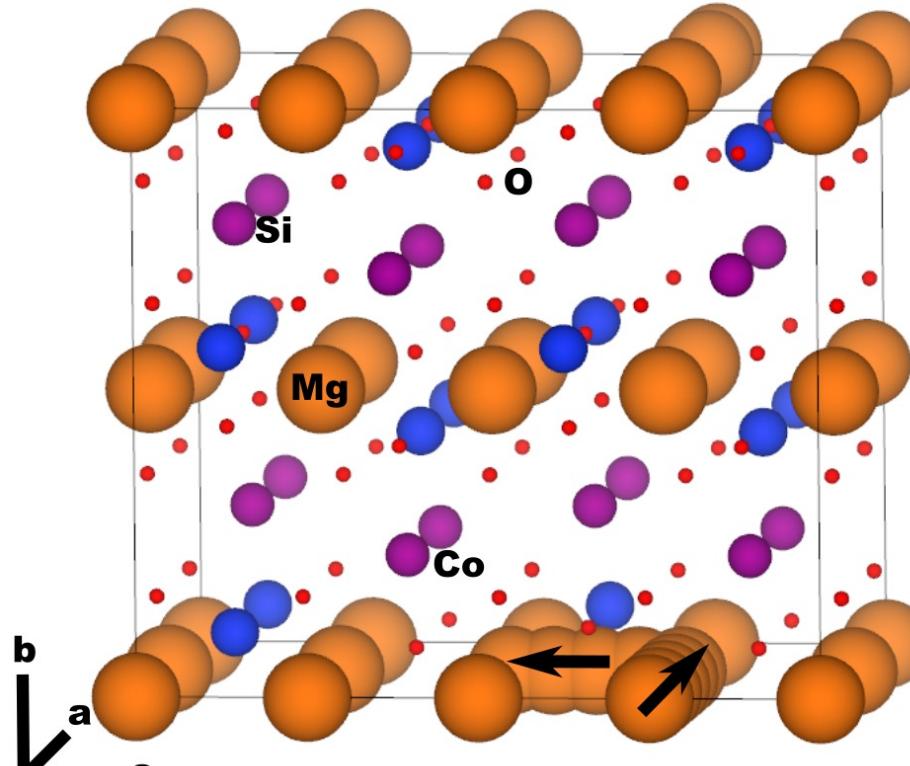
Careful split of dataset



559 datapoints in train; 60 in test

- Similar distribution in train and test
- Prevent unfair penalization: 1 test datapoint for groups contributing 1-2%
- Random sampling within each structural group

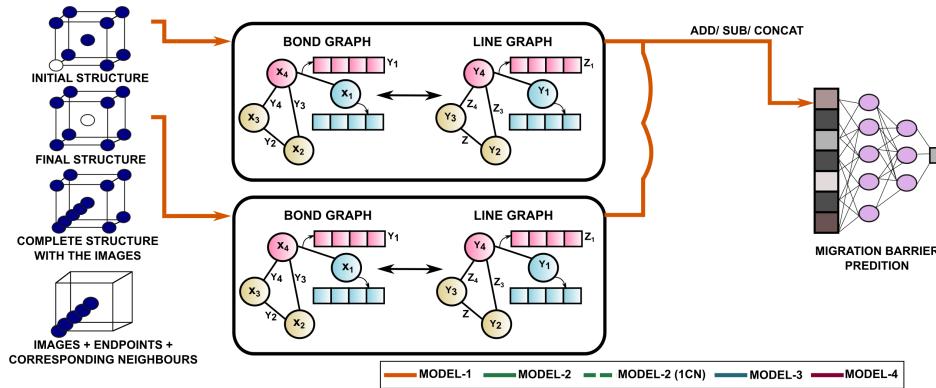
Distinguish multiple paths in a structure: use modified model architectures



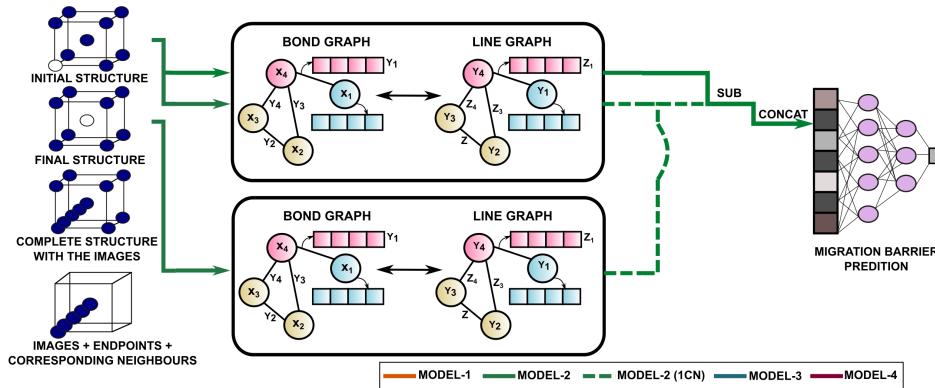
MgMnSiO_4 with two different pathways in the same structure

Distinguish multiple paths in a structure: use modified model architectures

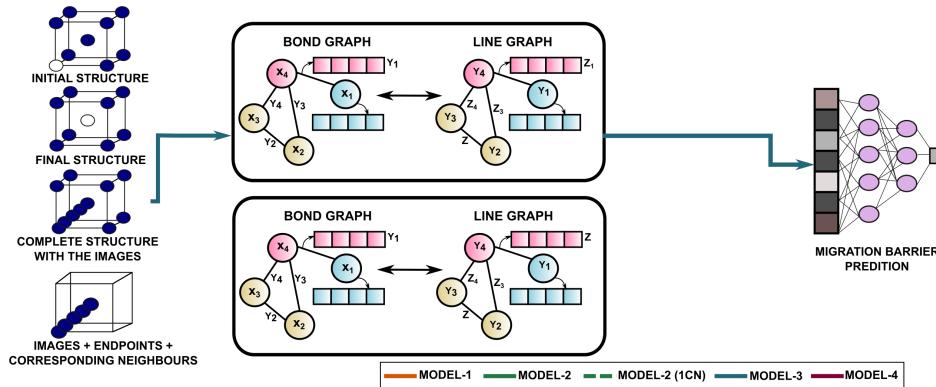
Model 1: Take initial and final as input



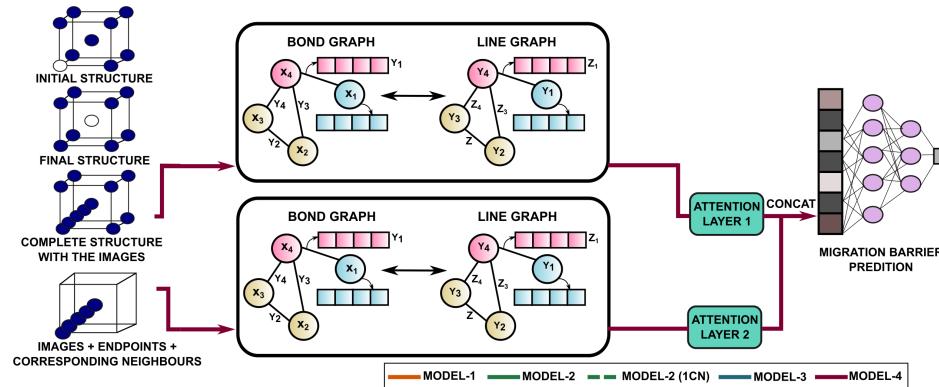
Model 2: Take initial and delta as input



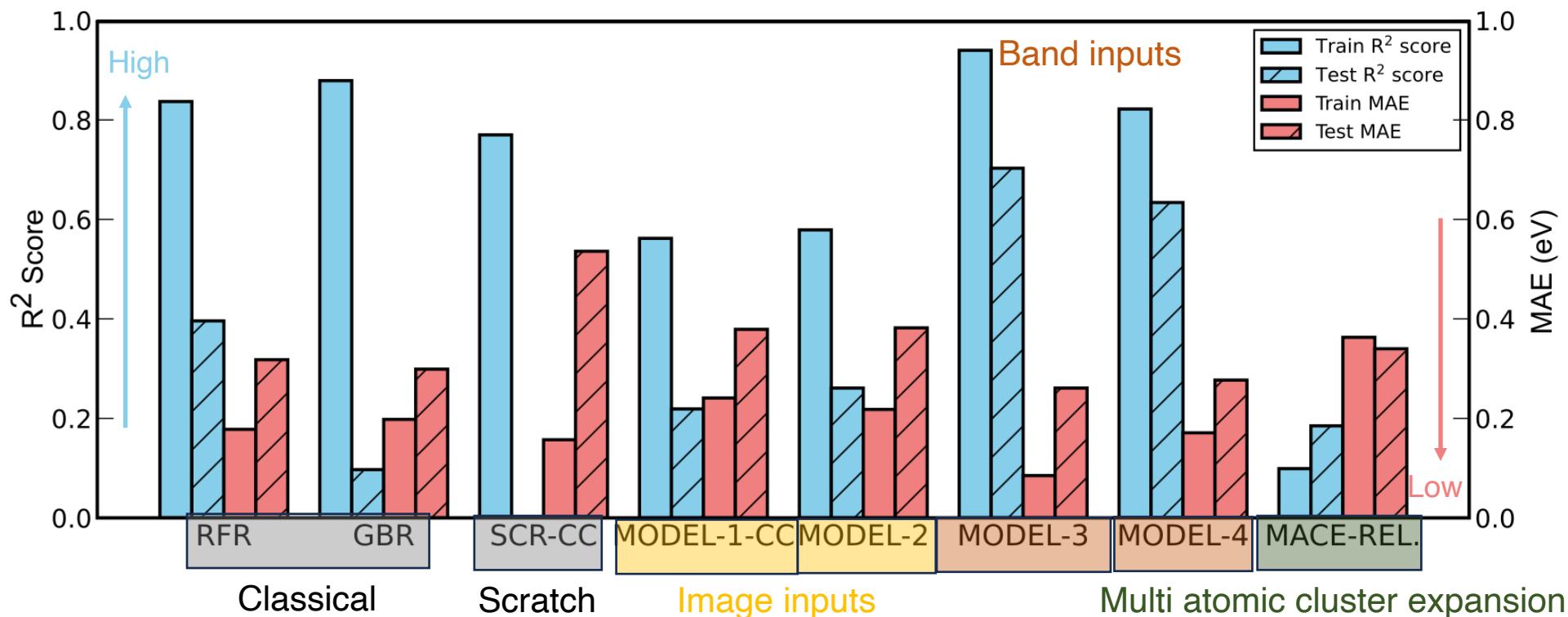
Model 3: Band as input



Model 4: Band with attention



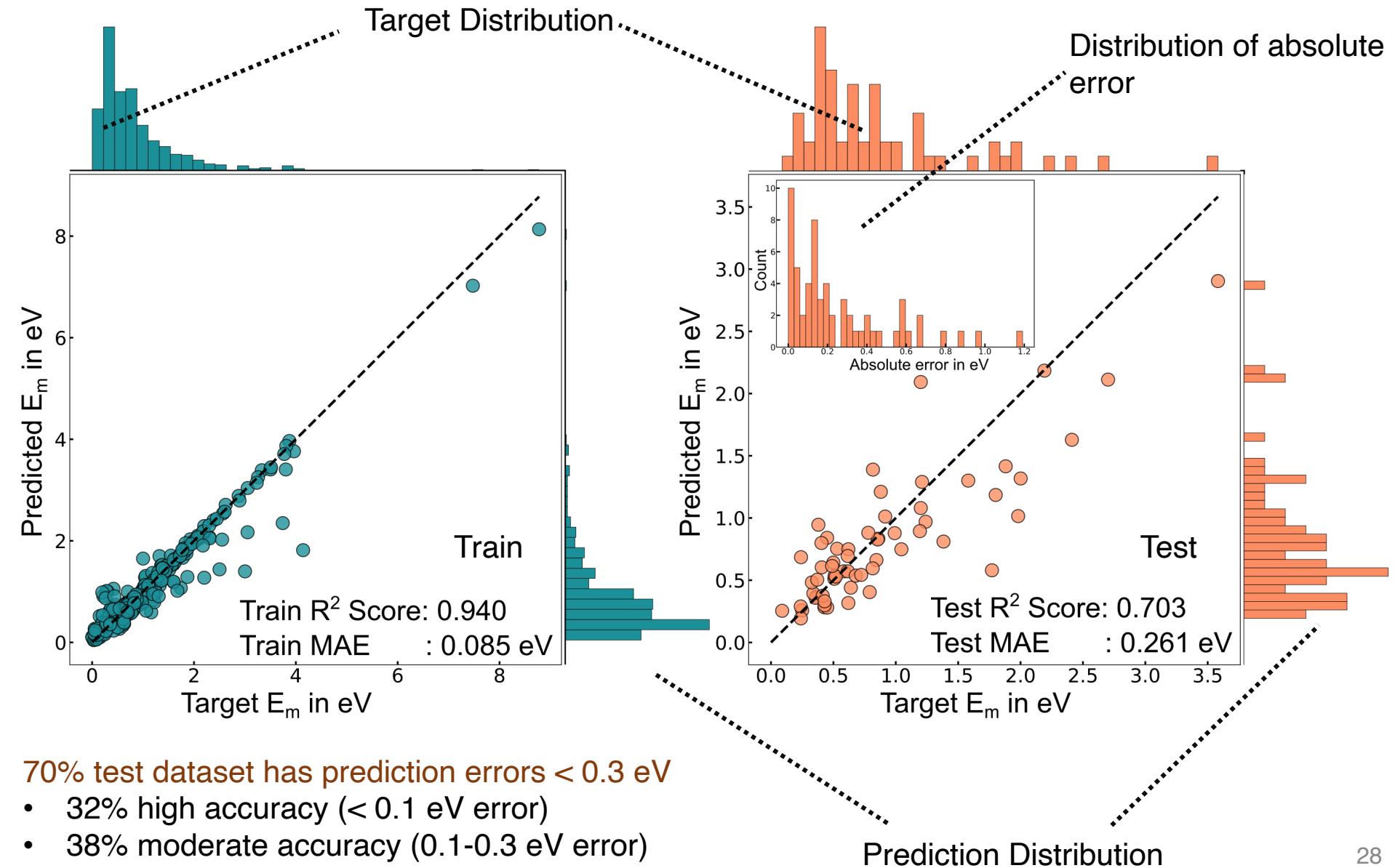
Model-3: best performing model



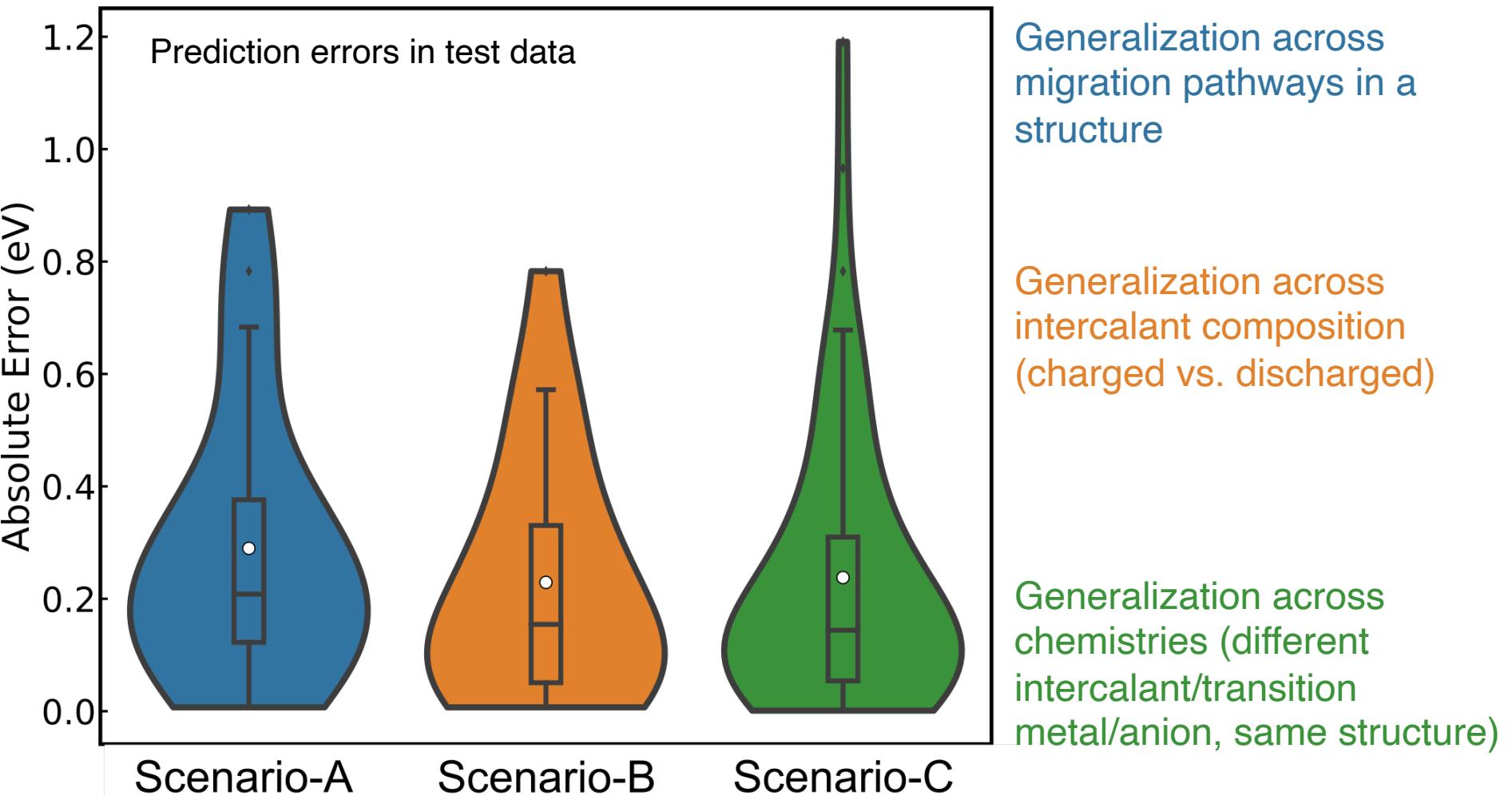
Model-3 performance (test set statistics, hashed bars):

- R²: 0.703, MAE: 0.261 eV (~1.4-1.8× DFT error)
- Better scores than classical and scratch models
- Versus MACE: slightly better MAEs, significantly better R²
- Identifies multiple migration pathways in a structure (band inputs > image inputs)
- Model-4 is closest performer to Model-3

Model-3 performance



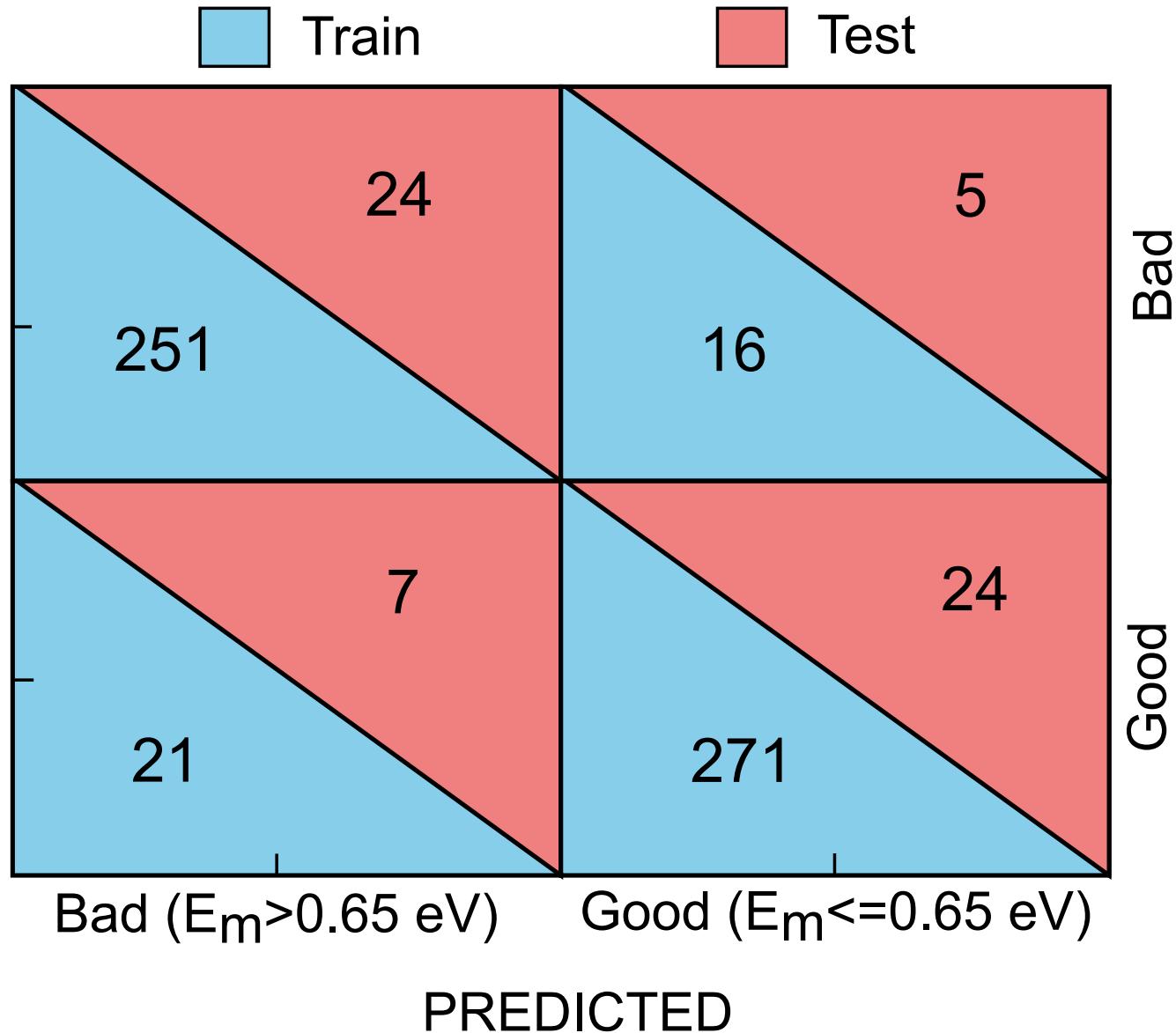
How does Model-3 generalize?



Lower mean/median errors for Model-3 in scenarios B & C: better generalization across composition and chemistry compared to migration pathways!

- Can be used as a screening tool

Model 3 as a classifier



Threshold: 0.65 eV

- Nano-sized cathode
- C/2 rate, 300 K

Accuracy: 80%

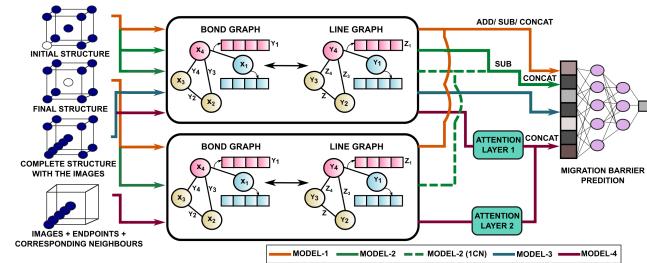
Precision:
82.8% (good conductors)
77.4% (bad conductors)

Model-3 can be good classifier!

- Especially for identifying good conductors

Conclusions

- Predicting E_m : critical for discovery of new materials with ‘high’ intercalant mobilities
- Accuracy: DFT gets qualitative trends right, reasonably close to experiments
- Dataset: curated 619 datapoints of calculated E_m
- Generalizable model to predict E_m
 - Leverage transfer learning, modified graph networks



“A literature-derived dataset of migration barriers for quantifying ionic transport in battery materials”, R. Devi, A. Balasubramanian, K.T. Butler, and G. Sai Gautam, [arXiv](#), 2508.06459 (2025) [under review].

“Optimal pre-train/fine-tune strategies for accurate material property predictions”, R. Devi, K.T. Butler, and G. Sai Gautam, [npj Comput. Mater.](#) 10, 300 (2024).

“Leveraging transfer learning for accurate estimation of ionic migration barriers in battery materials ”, R. Devi, K.T. Butler, and G. Sai Gautam, [arXiv](#), 2508.06436 (2025). [under review]